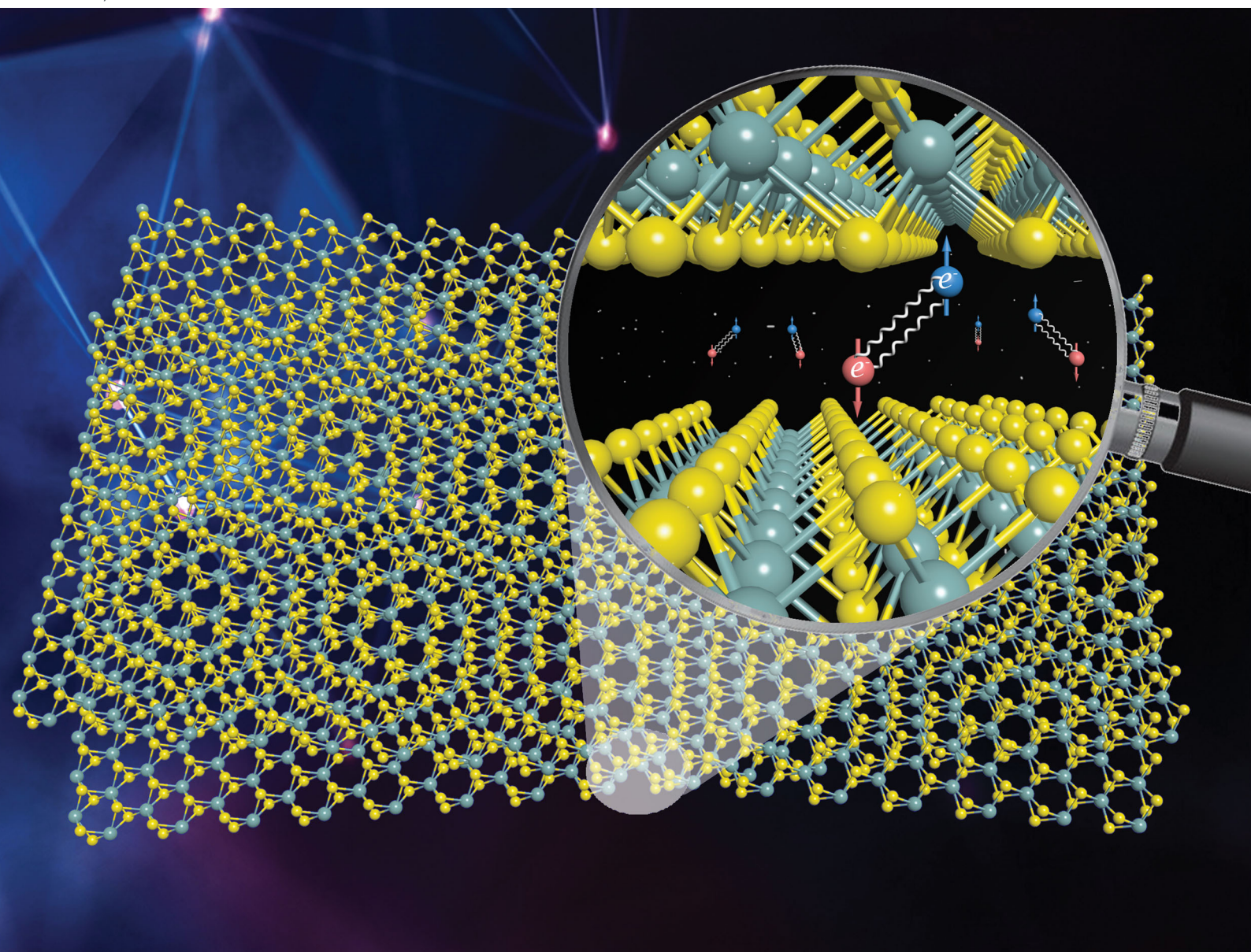


Nanoscale Horizons

The home for rapid reports of exceptional significance in nanoscience and nanotechnology

rsc.li/nanoscale-horizons



ISSN 2055-6756



Cite this: *Nanoscale Horiz.*, 2020, 5, 1309

Received 29th April 2020,
Accepted 16th July 2020

DOI: 10.1039/d0nh00248h

rsc.li/nanoscale-horizons

Interaction effects and superconductivity signatures in twisted double-bilayer WSe₂†

Liheng An,^{‡a} Xiangbin Cai,^{ib} Ding Pei,^b Meizhen Huang,^{ib} Zefei Wu,^a Zishu Zhou,^a Jiangxiazhi Lin,^a Zhehan Ying,^a Ziqing Ye,^a Xueming Feng,^a Ruiyan Gao,^a Cephise Cacho,^c Matthew Watson,^c Yulin Chen^b and Ning Wang^{ib} *^a

Twisted bilayer graphene provides a new two-dimensional platform for studying electron interaction phenomena and flat band properties such as correlated insulator transition, superconductivity and ferromagnetism at certain magic angles. Here, we present experimental characterization of interaction effects and superconductivity signatures in p-type twisted double-bilayer WSe₂. Enhanced interlayer interactions are observed when the twist angle decreases to a few degrees as reflected by the high-order satellites in the electron diffraction patterns taken from the reconstructed domains from a conventional moiré superlattice. In contrast to twisted bilayer graphene, there is no specific magic angle for twisted WSe₂. Flat band properties are observable at twist angles ranging from 1 to 4 degrees. Our work has facilitated future study in the area of flat band related properties in twisted transition metal dichalcogenide layered structures.

1. Introduction

The discovery of exotic superconducting phases in twisted bilayer graphene¹ has led to an increased interest in the study of modulated flat band properties and correlation effects in two-dimensional (2D) systems. Artificially stacking one single-layer graphene with respect to another flake of single-layer graphene at a small specified rotation angle, the so-called 'magic' angle, generates a moiré pattern or moiré unit-cells and reshapes the low energy electronic states, resulting in new flat bands and localized states. These flat bands have been proven to host exotic quantum phenomena such as

New concepts

The discovery of superconductivity, exotic insulating phases and ferromagnetism in twisted bilayer graphene has led to an increased interest in the study of modulated flat band properties and correlation effects in two-dimensional (2D) systems. To date, these unusual flat band properties have primarily been observed in graphene-graphene moiré superlattices. Recently, we first observe flat band states and superconductivity signatures in p-type semiconductor twisted double-bilayer WSe₂. We demonstrate that enhanced interlayer interactions can be observed when the twist angle decreases to a few degrees as reflected by the high-order satellites in the electron diffraction patterns taken from the 2H/AB'-stacked domains reconstructed from a conventional moiré superlattice. In contrast to twisted bilayer graphene, there is no specific magic angle for twisted WSe₂. The flat band properties are observed at twist angles ranging from 1 to 4 degrees. Twisted double-bilayer WSe₂ shows superconductivity signatures. With strong intrinsic electron-electron interaction effects and large effective masses, atomically thin twisted WSe₂ provides a new platform for studying flat band related properties and correlated behavior of 2D electrons.

Mott-insulator transition,² unconventional superconductivity and ferromagnetism.^{3–7} Similar flat band properties have also been evidenced in twisted double-bilayer graphene^{8–10} and in ABC stacked trilayer graphene moiré superlattices formed on hexagonal boron nitride (h-BN) sheets.¹¹ The dimension of the moiré superlattice is determined by the crystal lattice parameters and twisting angles. For a small twisting angle, the moiré pattern produces a long wavelength periodic modulation at the twisting interface and thus modifies the electronic band structure of the crystal structure through the formation of new low-energy sub-bands. Fabrication of twisted van der Waals heterostructures by mechanical exfoliation techniques offer new possibilities for 2D material band engineering.

To date, these unusual flat band characteristics^{12–14} have primarily been observed in graphene-graphene moiré superlattices created at a precisely controlled small twist angle (about or smaller than 1°) in order to enhance the electron-electron Coulomb interaction effects. Having a geometry similar to that of graphene, atomically thin semiconducting transition metal

^a Department of Physics, Center for Quantum Materials and William Mong Institute of Nano Science and Technology, the Hong Kong University of Science and Technology, Clear Water Bay, Hong Kong, China. E-mail: phwang@ust.hk

^b Department of Physics, Clarendon Laboratory, University of Oxford, Parks Road, Oxford OX1 3PU, UK

^c Diamond Light Source, Harwell Campus, Didcot OX11 0DE, UK

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0nh00248h

‡ These authors contributed equally to this work.

dichalcogenides (TMDCs) are potential candidates for fabricating 2D twisted heterostructures and exploring their transport and optical properties^{15,16} governed by flat bands and correlation effects. Recent studies have indicated that ultra-flat bands and interaction effects could be realized by moiré quantum well structures in the valence band of twisted bilayer TMDCs.^{17–22} TMDCs have large effective masses, relatively strong electron–electron interactions^{23–25} and more additional effects stemming from the lack of inversion symmetry and large spin–orbit interactions. On the other hand, TMDCs have advantages for the exploration of the flat band properties over a continuous range of twist angles. Because the electronic structure is easily tunable, this material system provides an ideal platform for exploring other emergent electronic states, for example, the twisted angle dependent magnetic ordering, current-induced magnetic order switching, non-linear Hall effects, *etc.* Besides TMDCs, other ultrathin 2D nanostructures, acting as building blocks, can be used to build up new twist bilayer nanostructures or even multiple layers with designed artificial structures.²⁶ Here, we present experimental results demonstrating the successful fabrication of twisted double-bilayer WSe₂ device structures with enhanced interlayer interactions as observed by atomic resolution imaging of the reconstructed domains from a conventional moiré superlattice and interesting transport characteristics measured at cryogenic temperatures. Our angle-resolved photoemission spectroscopy (ARPES) measurement reveals the formation of the four-band valleys at the Γ point in the Brillouin zone and indicates strong interaction effects occurring between WSe₂ bilayers. Distinct from graphene, the flat band properties are observable in WSe₂ at relatively wide twist angles ranging from 1° to 4°, offering new design opportunities for fabricating moiré quantum well heterostructures in TMDCs.

2. Device fabrication

The encapsulated twisted double bilayer WSe₂ devices are fabricated by a dry transfer method.^{27,28} Bilayer WSe₂ flakes are first exfoliated on the surfaces of 300 nm-thick SiO₂/Si substrates and picked up by another flake of h-BN. The crystal orientation of bilayer WSe₂ is determined by its natural edges. Then, another bilayer WSe₂ flake with straight edges is selected and picked up by the first flake of WSe₂ at a specific twisting angle (4° for device D11, 1° for device E7 and 2° for device F²). Finally, the whole stacked structure is transferred to another flake of bottom h-BN with pre-patterned Cr/Pt (5 nm/15 nm) bottom electrodes. We are able to control the twist angle between different layers. Without annealing, the accuracy of the twisted angles is about ± 0.2 degree. After annealing, however, there is generally a deviation from the designed twist angles ($< 0.4^\circ$). This transfer process is conducted in a glove box in which O₂ and H₂O concentrations are less than 1 ppm. Annealing treatment is performed under Ar/H₂ protecting gas atmosphere at 250 °C to ensure the good contacts between WSe₂ and Pt bottom electrodes. Hall-bar shaped Ti/Au (5 nm/60 nm)

is deposited on the top surface of upper h-BN to form the top gate, covering the area of the twisted double bilayer WSe₂. All transport measurements are performed using standard lock-in technique at cryogenic temperatures. Nano-ARPES measurements are performed at the Beamline I05 of the Diamond Light Source, UK. The sample is annealed at 250 °C for 8 hours before measurement. All ARPES measurements are carried out under ultrahigh vacuum ($< 1 \times 10^{-10}$ mbar) at 50 K. The spatial resolution for locating the sample is better than $2 \times 2 \mu\text{m}^2$. The overall energy and angle resolution are 30 meV and 0.25°, respectively.

3. Results and discussion

Fig. 1a and b illustrate the device structure of the twisted double-bilayer WSe₂ fabricated by the exfoliation and dry transferring techniques.^{27,28} The twisted double-bilayer WSe₂ channel is controlled in a double-gating configuration for this study. Ohmic contacts (Pt bottom electrodes²⁹) are achieved by applying a large negative bias *via* the top-gate, which enables the carrier density to be tuned at the same time. Fig. 1c is a schematic moiré superlattice formed by twisted double-bilayer WSe₂ flakes. Fig. 1d is the atomic model of 2H-stacked (ABAB) and AB'-stacked double-bilayer WSe₂ viewed along the $\langle 100 \rangle$ zone axis. The crystallographic structures of the twisted double-bilayer WSe₂ are investigated by high-resolution scanning transmission electron microscopy (STEM). Using the same transferring and stacking techniques, we mechanically transfer the double-bilayer WSe₂ flakes with desired twist angles onto lacey carbon grids for STEM study. As shown in the selected-area electron diffraction patterns in Fig. 1e, the 1.1° twist angle between the two sets of hexagonal diffraction patterns is evident which is close to the designed angle of 1°. Theoretically, overlapping two layers of a crystalline structure at a small twist angle generates a moiré pattern as schematically depicted in Fig. 1c. In this case, one should observe two sets of hexagonal diffraction patterns (the Bragg diffractions). As illustrated in the enlarged pictures of certain diffractions such as (010), (110) and (020) in Fig. 1f–h, however, we observe high-order satellites (from first to third orders) surrounding each pair of these Bragg diffractions, indicating that strong interlayer interactions or interface reconstructions occur in the twisted double-bilayer WSe₂. According to the reciprocal spacing of the satellite diffractions, the unit-cell parameter of the newly formed moiré pattern is about 20–25 nm.

As verified by aberration-corrected annular dark-field imaging in STEM, the twisted WSe₂ bilayers undergo a structural reconstruction and form alternately distributed triangular regions as marked in blue and red in Fig. 1i. This structural reconstruction leads to reshaping of the ideal hexagonal moiré pattern (Fig. 1c) to a triangular pattern (Fig. 1i). A detailed model of such a reconstruction and formation of triangular shapes can be found in ref. 30. For a small twist angle, the area of stacking sequence AB' at the interface between the bilayer WSe₂ (as shown schematically in Fig. 1d) extends to form the triangular shape.

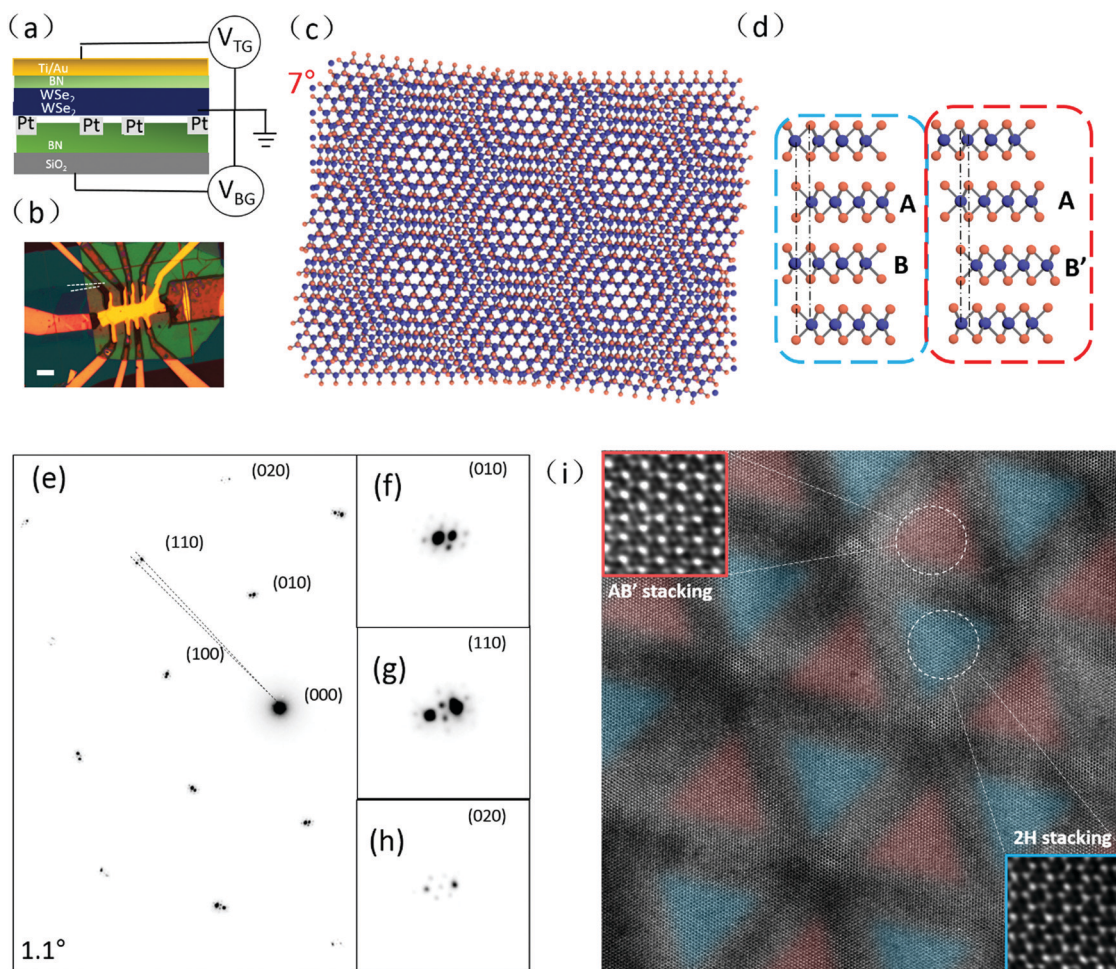


Fig. 1 Device structure, stacking model and atomic-scale reconstruction in twist double bilayer WSe₂. (a) and (b) Schematic and optical images of the encapsulated twisted double-bilayer WSe₂. The scale bar is 6 μm. (c) A schematic moiré superlattice formed by twisted double-bilayer WSe₂ flakes. (d) Atomic model of 2H-stacked (ABAB) and AB'-stacked double-bilayer WSe₂ viewed along the $\langle 100 \rangle$ zone axis. (e) Electron diffraction pattern taken from a twisted double-bilayer WSe₂ with enlarged views of respective Bragg spots (f)–(h), showing the high-order satellites caused by interface reconstruction. (i) Annular dark field image of the interface-reconstructed region, in which 2H/AB'-stacked domains are marked in blue/red. The zoomed-in images illustrate the 2H and AB' atomic structures.

These 2H and AB' domains are separated by the triangular network boundaries (the dark contrast strips of about ~ 3 nm in width). The narrow domain boundaries have lattice misfit strain as reflected by the dark image contrast. However, the image contrast inside each blue or red area is uniform, meaning that the lattice misfit strain has been relaxed. The scale of the triangular pattern is consistent with the reciprocal space measurement of the electron diffraction pattern. The atomic-resolution images (the insets in Fig. 1i) reveal that 2H and AB' stacking orders³¹ of WSe₂ exist in the blue and red regions respectively, further verified that the structural reconstruction from a conventional moiré pattern occurs and produces a well-relaxed stacking structure.

The band structure of WSe₂ is layer number dependent. In monolayer and bilayer WSe₂,³² the valence band maxima locate at K -valleys. In a few-layer WSe₂ with a layer number greater than three, Γ -valley heavy holes²³ dominate the carrier transport in the valence band edge. In a four-layer WSe₂ device, the

electrical conductance measured at cryogenic temperatures (see Fig. 2a) monotonically increases with increasing the gate voltage or carrier concentration. In twisted double-bilayer WSe₂ (device D11), however, the conductance normally exhibits a few peaks and dips with increasing the gate voltage (see Fig. 2a for the 4° case), indicating the existence of localization states resulted from the moiré superlattice structures. As an example, Fig. 2c illustrates the measured longitudinal resistance of the sample with a 4° twist angle at different gating configurations. By setting the top gate V_{TG} to -22 V (through a 38 nm h-BN layer shown in Fig. S3, ESI†), the channel resistance measured by a four-lead configuration is about 200 Ohms and the corresponding carrier density is $7.5 \times 10^{12} \text{ cm}^{-2}$. The hole mobility calculated by $\mu_h = \frac{Ld\sigma}{Wdn}$. Considering $\frac{L}{W} = 1/3$, the field effect transistor mobility has reached $1000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, which is slightly lower than the high performance devices of monolayer and bilayer WSe₂ reported previously.^{24,33}

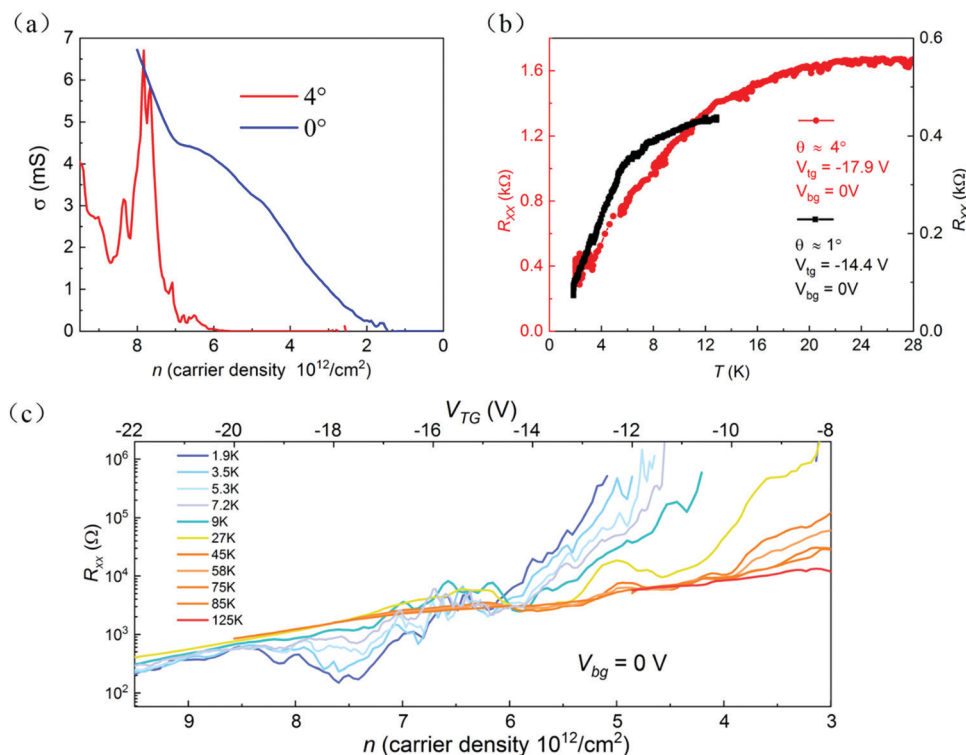


Fig. 2 Experimental data of superconductivity obtained from twisted double bilayer WSe₂. (a) Conductance of intrinsic four layers WSe₂ and twisted double-bilayer WSe₂ devices. (b) Quick resistance reduction of the 1° and 4° twisted devices at low temperatures. (c) Longitudinal resistance plotted as a function of the carrier density at different temperatures. The excitation $V_{ds} = 1$ mV and back gate $V_{BG} = 0$ V.

For the twisted double-bilayer (totally 4 layers) device, the valence band edges locate at the Γ -valley with a two-fold degeneracy. The nano-ARPES measurement confirmed that the Γ -valley is slightly higher (~ 80 meV) than the K -valley (Fig. 3c),

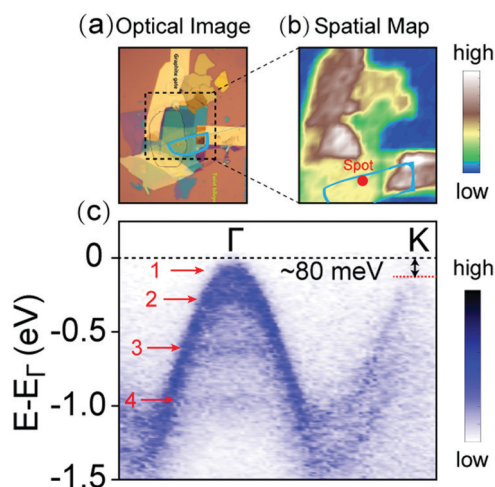


Fig. 3 ARPES data obtained from device A3 (twist angle 3.5°). (a) The optical image of the device. The twisted double bilayer area is marked by the blue circle. (b) Scanning photoemission microscopy image of the device. The intensity of the map represents the counts of photoelectron near E_F . (c) electronic band dispersion along Γ - K points to show the four-band Γ -valley characteristics of the twisted double bilayer WSe₂ area. The ARPES measurement was performed at the red spot marked in (b).

which is consistent with that of a previous layer-dependent ARPES study of WSe₂.³⁴ Given that, the two peaks shown in the longitudinal resistance at $V_{TG} = -15.4$ V and -19.8 V in Fig. 2c, could then correspond to the scattering effect of moiré lattice at $6.5 \times 10^{12} \text{ cm}^{-2}$ and at $9 \times 10^{12} \text{ cm}^{-2}$ respectively based on the following estimation. The moiré pattern/superlattice constant^{35,36} is expressed by $\lambda = a / \left(2 \sin \left(\frac{\theta}{2} \right) \right)$, where a is the lattice constant of WSe₂. λ is estimated to be 4.7 nm for a 4° twist angle. The full filling of the moiré superlattice (2 holes per moiré unit cell) can be worked out as $n_0 = 2 / \left(\frac{\sqrt{3}}{2} \lambda^2 \right) \approx 1.05 \times 10^{13} \text{ cm}^{-2}$. This

value is quite close to the carrier concentration calculated according to the quantum capacitance of the WSe₂-BN heterostructure. In principle, there are two metallic states adjacent to the correlated half filling insulating states and band insulating states. While here, no insulating states are observed and we only find metallic states with a few bumps and dips in the four-end resistance. As shown in Fig. 2c, the first one locates at $V_{TG} = -14.8$ V and displays metallic characteristics since the resistance measured at this gate voltage decreases with decreasing sample temperature from 125 K to 10 K. While, it does not change so much below 10 K. The second one locates between $V_{TG} = -16$ V to -19.6 V with carrier densities ranging from 5.9 to $8.3 \times 10^{12} \text{ cm}^{-2}$. It has a metallic behavior in the temperature region of 125–11 K. The measured resistance decreases gradually with decreasing sample temperature,

which means that phonon scattering dominates in the electron transport behavior. From 11 K to 1.9 K, however, a quick reduction of the resistance (from 1600 to 200 Ohms) occurs as shown in Fig. 2b. Such a quick reduction of the channel resistance is possibly attributed to the formation of superconducting states at the interface of twisted double-bilayer WSe₂. Because of the impurity-dominated scattering in WSe₂, the intrinsic resistance of a few-layer WSe₂ device normally remains almost as a constant value at very low temperatures,³⁷ and has no contribution to the observed quick reduction of the channel resistance.

The flat bands formed at the edges of the valence band in WSe₂ moiré superlattice have been predicted theoretically.^{17,18,21} The bandwidth is comparable to that of twisted graphene at the 'magic' angle. In these flat bands, the kinetic energy is quenched ($E_k \sim 10$ meV) which is in the same order as the on-site Coulomb interaction $U \sim \frac{e^2}{\epsilon\lambda}$, where $\epsilon = 4$ is the dielectric constant of h-BN. For $\lambda = 4.7$ nm, the Coulomb interaction is about dozens of meV. This could lead to the superconducting states at the interface of twisted double-bilayer WSe₂. By fixing the carrier density at $7.5 \times 10^{12} \text{ cm}^{-2}$, we observe that the longitudinal resistance decreases quickly when sample temperature is below 18 K (Fig. 2b). This transition temperature is much higher than the highest onset transition temperature of twisted graphene. At 6 K, the channel resistance drops to 800 Ω , about half of the normal metallic state resistance.

To investigate the electronic band structure of the twisted double bilayer WSe₂, we have performed state-of-the-art nano-ARPES measurement on the devices without top gates. Fig. 3a presents the optical image of the devices used in the ARPES measurement. The size of the twisted area (marked by the blue circle) is around $10 \times 20 \mu\text{m}^2$, which is sufficiently large compared with the ARPES spot size. Fig. 3b is the scanning photoemission microscopy image of one device, in which the intensity represents the counts of photoelectrons near the Fermi level. Thus, the gold pad and the graphite gate area have a relatively higher intensity than the WSe₂ area. By comparing Fig. 3a and b, the twisted area can be located for the ARPES measurement. The electronic band dispersion along Γ -K points is shown in Fig. 3c. At the Γ point, we can identify four bands near the valence band top, which indicates strong interlayer coupling in the twisted double bilayer system. As the number of the bands at Γ -valley in WSe₂ is determined by the layer number, the observation of four-band Γ -valley suggests that the electronic structure of the twisted double bilayer system is close to the intrinsic 4-layer system. This means the electronic states in top and bottom bilayers have been unified into an integrated system by the interlayer interaction. Furthermore, we find that the Γ -valley is slightly higher (~ 80 meV) than the K-valley, which implies that the Γ valley have more contribution to the electrical properties in the p-doped region. This discovery distinguishes this sample system from twisted graphene or twisted bilayer WSe₂ whose valence band maxima locate at the boundaries of the Brillouin zone.

The transport characteristics of the twisted WSe₂ show a similar behavior of superconductivity in twisted graphene structures.

Our finding suggests that the interaction effects induced by twisting bilayer WSe₂ are substantial. Due to the structural inhomogeneity in our twisted layer WSe₂, the resistance cannot reduce further at cryogenic temperatures. As observed by atomic-resolution electron microscopy, the twist angle often slightly changes from area to area, resulting in micrometer-sized domains in the sample. Because our device size is about a few micrometers, the electrodes for transport measurement may cross more than one domain or more than one superconducting domain if superconducting state exists in each domain or in some domains.³⁸ The reason why twisted WSe₂ have a high-density of domains is unclear. This could be introduced by fabrication or originated from the intrinsic structure properties of WSe₂ since in each layer of WSe₂ there are three atomic layers of tungsten and selenium atoms. The atomic bonds in WSe₂ are weaker than the sp^2 bonds in graphene. This may result in slight deviation of the twist angle by locally non-uniform strains at the interface.

A small twist angle leads to the formation of a large moiré superlattice cell. For E7 device (1° twist angle and $\lambda \approx 19$ nm),

fully filling the moiré superlattice requires $n_0 = 2 \left/ \frac{\sqrt{3}}{2} \lambda^2 \right. \approx$

$0.65 \times 10^{12} \text{ cm}^{-2}$, an order of magnitude smaller than that of the 4° twist sample. Similarly, by changing the top-gate voltage, we observe multiple superconducting-like states in the 1° twist device (Fig. 4a), which could be resulted from the high-order flat bands. The thickness of the h-BN dielectric layer in this device is about 44 nm (shown in Fig. S6, ESI[†]). The separation between the first and second superconducting-like domes is $V \approx 1$ V, corresponding to a carrier density of $0.66 \times 10^{12} \text{ cm}^{-2}$, which is very close to the full filling state of the moiré superlattice unit cell shown in Fig. 1i. In Fig. 2b, we find that the resistance remains nearly a constant value above 5 K, and a quick resistance drop occurs below 5 K.

We suggest that the multiple superconducting-like domes shown in Fig. 4a are attributed to the high-order flat bands. Because of structural inhomogeneity in the twisted structures, the domes do not occur periodically. Fig. 4b illustrates the longitudinal resistance of another small angle of 2° twist sample. Multiple bumps and dips are clearly visible. In addition to the similar thermal effects on the superconductor states, we observe that a perpendicular magnetic field can change the superconducting-like states to normal metallic states. In Fig. 4c, the longitude resistance does not change when the magnetic field is above 0.3 T, which means the system become a normal metallic state if the magnetic field is higher than 0.3 T.

Fig. 5a shows the transport characteristics of device F² (twist angle $2 \pm 0.5^\circ$), in which the longitude resistance is measured by setting the top gate at -6.65 V and -6.92 V. The carrier density is estimated to be around $n_0 = 6 \times 10^{12} \text{ cm}^{-2}$ and $7 \times 10^{12} \text{ cm}^{-2}$. Considering the full filling states of $n_0 = 2.6 \times 10^{12} \text{ cm}^{-2}$ for a 2° twist angle, we attribute that the electrons possibly reside on the high order flat bands^{18,39} of the Γ -valley in this sample. For $V_{\text{TG}} = -6.65$ V, the resistance keeps a constant value from 30 K to 9 K and quickly drops down from

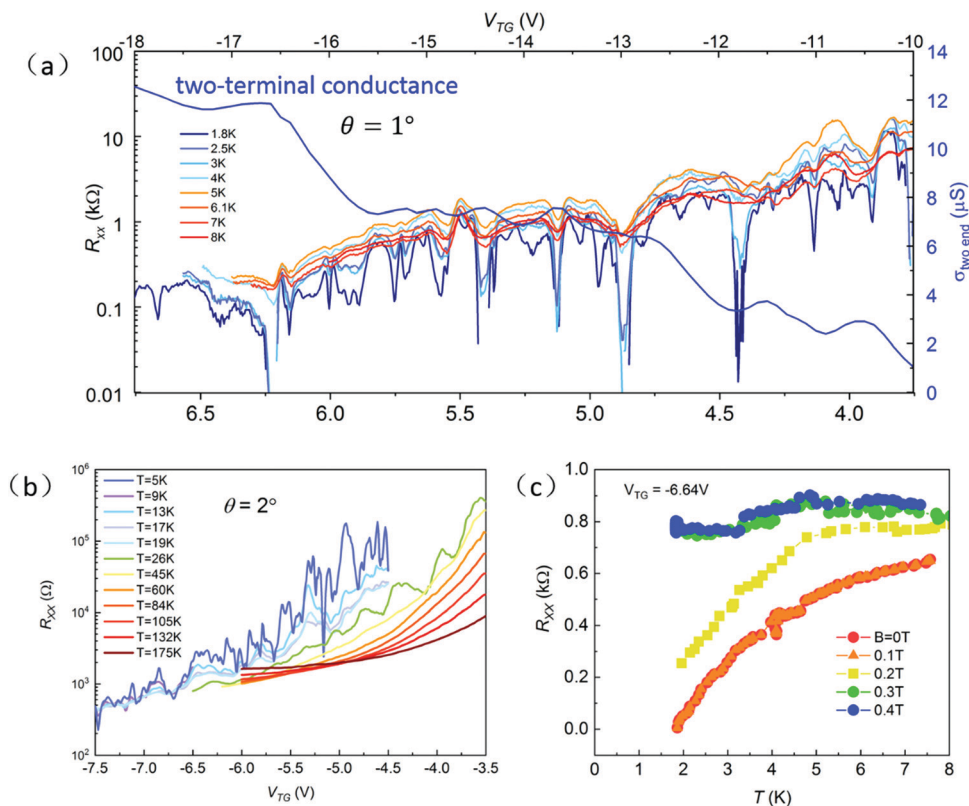


Fig. 4 Transport characteristics in 1°, 2° twisted double bilayer WSe₂. (a) The resistance (four terminal measurement) of Sample E7 with a twisting angle of about 1° at various temperatures (1.9 K to 8 K). The right blue line is the two terminal conductance of the same sample. (b) The resistance (four terminal measurement) of Sample F² with a twisting angle of about 2° at various temperatures (1.9 K to 8 K). (c) The longitude resistance of Sample F² at $V_{TG} = -6.64$ V, plotted as a function of temperature with various perpendicular magnetic fields.

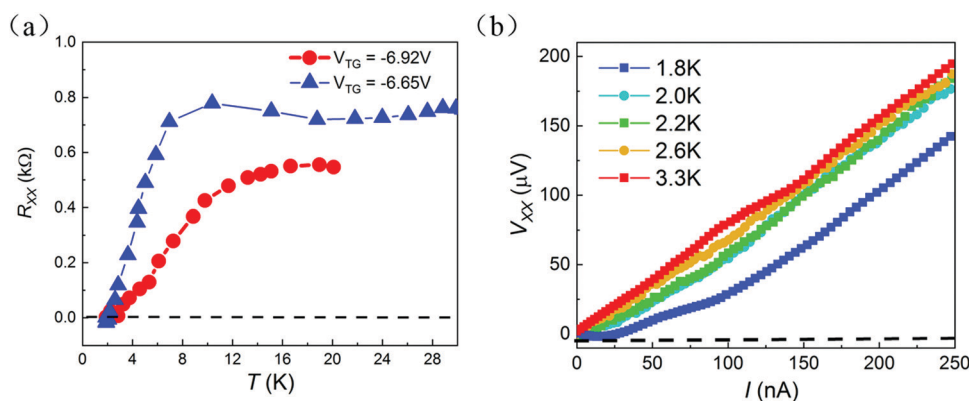


Fig. 5 Superconductivity characteristics in 2° twisted double bilayer WSe₂. (a) Longitudinal resistance measured by the four-terminal configuration of the 2° twist device. (b) The I - V curves measured from device F² at different temperatures. The V_{TG} is set at -6.65 V.

9 K to 2 K. It reaches a zero resistance with an uncertainty of ± 10 Ohms. At $V_{TG} = -6.92$ V, the resistance keeps a constant value from 20 K to 10 K and quickly drops down from 10 K to 2 K to reach a zero resistance at about 3 K. Fig. 5b demonstrates a typical non-linear I - V curve observed at $V_{TG} = -6.64$ V. At 1.8 K, the critical current for hosting the zero resistance is about 25 nA. According to the experimental data collected from the samples with different twist angles, we confirm that enhanced interlayer interactions which are similar to that in

twisted graphene are generated. Because of the structural inhomogeneity in the twisted WSe₂ devices, precisely measuring the superconductivity parameters, for example, the critical current density undergoes difficulties. This is because the induced electric field *via* V_{ds} across the inhomogeneous domains and domain boundaries cannot be well defined. On the other hand, the van der Waals interface structure in the twist devices may be in an unstable state since annealing of the devices very often causes the twist angle changes. However, our experimental

results qualitatively demonstrate the superconductivity signatures in twisted WSe₂ structures.

In conclusion, we successfully fabricated double-bilayer twisted WSe₂ devices with well-controlled twist angles. Reconstructed moiré superlattices are observed in the samples with 1° twist angle. WSe₂ moiré lattices can host full filling carrier densities ranging from $0.65 \times 10^{12} \text{ cm}^{-2}$ to $1.05 \times 10^{13} \text{ cm}^{-2}$. Enhanced interlayer interactions are observed when the twist angle decreases to a few degrees as reflected by the high-order satellites in the electron diffraction patterns taken from the reconstructed domains from a conventional moiré superlattice. We demonstrate transport characteristics with superconductivity signatures measured at cryogenic temperatures which are modulated by the reconstructed moiré superlattices and inhomogeneous domains. With strong intrinsic electron–electron interaction effects and large effective masses, atomically thin twisted WSe₂ provides a new platform for studying the correlated phenomena of 2D electrons.

Author contributions

LHA fabricated twist double bilayer WSe₂ devices and TEM samples. LHA, JXZL, ZFW, HMZ performed and analyzed the transport measurements under supervision of NW. LHA, ZFW, HMZ and NW analyze the quantum transport data. XBC took and analyzed the HRTEM image for the twist WSe₂. DP performed the ARPES experiment with the assistance of CC and MW. Other authors provided technical assistance on sample preparations and measurements and discussed on the paper. LHA and NW wrote the paper with input from all authors.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

We thank Prof. Xi Dai, Prof. K. T. Law, Prof. Nian Lin, Dr Jianpeng Liu and Dr Wenyu He, Dr Tong Zhou, Dr En Li, from HKUST for fruitful discussions. This work is supported by the Research Grants Council of Hong Kong (Project No. 16300717 and C7036-17W) and EPSRC (grant no. EP/M020517/1). We acknowledge the technical support from the Super-resolution Electron Microscopy facility and Raith-HKUST Nanotechnology Laboratory for the electron-beam lithography facility at MCPFP, and the support from Beamline I05, Diamond Light Source (proposal no. SI24366). Ding Pei thanks the support from Chinese Scholarship Council.

References

- 1 Y. Cao, V. Fatemi, S. Fang, K. Watanabe, T. Taniguchi, E. Kaxiras and P. Jarillo-Herrero, *Nature*, 2018, **556**, 43–50.
- 2 Y. Cao, V. Fatemi, A. Demir, S. Fang, S. L. Tomarken, J. Y. Luo, J. D. Sanchez-Yamagishi, K. Watanabe, T. Taniguchi, E. Kaxiras, R. C. Ashoori and P. Jarillo-Herrero, *Nature*, 2018, **556**, 80–84.
- 3 A. L. Sharpe, E. J. Fox, A. W. Barnard, J. Finney, K. Watanabe, T. Taniguchi, M. A. Kastner and D. Goldhaber-Gordon, *Science*, 2019, **365**, 605–608.
- 4 A. H. MacDonald, *Physics*, 2019, **12**, 12.
- 5 X. Lu, P. Stepanov, W. Yang, M. Xie, M. A. Aamir, I. Das, C. Urgell, K. Watanabe, T. Taniguchi and G. Zhang, 2019, arXiv preprint arXiv:1903.06513.
- 6 M. Serlin, C. Tschirhart, H. Polshyn, Y. Zhang, J. Zhu, K. Watanabe, T. Taniguchi, L. Balents and A. Young, *Science*, 2019, **8**.
- 7 W.-Y. He, D. Goldhaber-Gordon and K. Law, 2019, arXiv preprint arXiv:1908.11718.
- 8 X. Liu, Z. Hao, E. Khalaf, J. Y. Lee, K. Watanabe, T. Taniguchi, A. Vishwanath and P. Kim, 2019, arXiv preprint arXiv:1903.08130.
- 9 C. Shen, N. Li, S. Wang, Y. Zhao, J. Tang, J. Liu, J. Tian, Y. Chu, K. Watanabe and T. Taniguchi, 2019, arXiv preprint arXiv:1903.06952.
- 10 A. Kerelsky, C. Rubio-Verdú, L. Xian, D. M. Kennes, D. Halbertal, N. Finney, L. Song, S. Turkel, L. Wang and K. Watanabe, 2019, arXiv preprint arXiv:1911.00007.
- 11 G. Chen, L. Jiang, S. Wu, B. Lyu, H. Li, B. L. Chittari, K. Watanabe, T. Taniguchi, Z. Shi and J. Jung, *Nat. Phys.*, 2019, **15**, 237.
- 12 P. Lucignano, D. Alfè, V. Cataudella, D. Ninno and G. Cantele, *Phys. Rev. B*, 2019, **99**, 195419.
- 13 N. F. Yuan and L. Fu, *Phys. Rev. B*, 2018, **98**, 045103.
- 14 J. Liu, J. Liu and X. Dai, *Phys. Rev. B*, 2019, **99**, 155415.
- 15 K. L. Seyler, P. Rivera, H. Yu, N. P. Wilson, E. L. Ray, D. G. Mandrus, J. Yan, W. Yao and X. Xu, *Nature*, 2019, **567**, 66.
- 16 C. Jin, E. C. Regan, A. Yan, M. I. B. Utama, D. Wang, S. Zhao, Y. Qin, S. Yang, Z. Zheng and S. Shi, *Nature*, 2019, **567**, 76.
- 17 M. H. Naik and M. Jain, *Phys. Rev. Lett.*, 2018, **121**, 266401.
- 18 F. Wu, T. Lovorn, E. Tutuc, I. Martin and A. MacDonald, *Phys. Rev. Lett.*, 2019, **122**, 086402.
- 19 M. Fleischmann, R. Gupta, S. Sharma and S. Shallcross, arXiv preprint arXiv:1901.04679, 2019.
- 20 L. Wang, E.-M. Shih, A. Ghiotto, L. Xian, D. A. Rhodes, C. Tan, M. Claassen, D. M. Kennes, Y. Bai and B. Kim, arXiv preprint arXiv:1910.12147, 2019.
- 21 F. Wu, T. Lovorn, E. Tutuc and A. H. MacDonald, *Phys. Rev. Lett.*, 2018, **121**, 026402.
- 22 F. Conte, D. Ninno and G. Cantele, *Phys. Rev. B*, 2019, **99**, 155429.
- 23 S. Xu, J. Shen, G. Long, Z. Wu, Z. Q. Bao, C. C. Liu, X. Xiao, T. Han, J. Lin, Y. Wu, H. Lu, J. Hou, L. An, Y. Wang, Y. Cai, K. M. Ho, Y. He, R. Lortz, F. Zhang and N. Wang, *Phys. Rev. Lett.*, 2017, **118**, 067702.
- 24 H. C. P. Movva, B. Fallahazad, K. Kim, S. Larentis, T. Taniguchi, K. Watanabe, S. K. Banerjee and E. Tutuc, *Phys. Rev. Lett.*, 2017, **118**, 247701.
- 25 J. Lin, T. Han, B. A. Piot, Z. Wu, S. Xu, G. Long, L. An, P. Cheung, P. P. Zheng, P. Plochocka, X. Dai, D. K. Maude, F. Zhang and N. Wang, *Nano Lett.*, 2019, **19**, 1736–1742.

- 26 Y. Wang, C. Sun, X. Yan, F. Xiu, L. Wang, S. C. Smith, K. L. Wang, G. Q. Lu and J. Zou, *J. Am. Chem. Soc.*, 2011, **133**, 695–697.
- 27 S. Xu, Z. Wu, H. Lu, Y. Han, G. Long, X. Chen, T. Han, W. Ye, Y. Wu, J. Lin, J. Shen, Y. Cai, Y. He, F. Zhang, R. Lortz, C. Cheng and N. Wang, *2D Mater.*, 2016, **3**, 021007.
- 28 Z. Wu, S. Xu, H. Lu, A. Khamoshi, G.-B. Liu, T. Han, Y. Wu, J. Lin, G. Long and Y. He, *Nat. Commun.*, 2016, **7**, 12955.
- 29 H. C. Movva, A. Rai, S. Kang, K. Kim, B. Fallahazad, T. Taniguchi, K. Watanabe, E. Tutuc and S. K. Banerjee, *ACS Nano*, 2015, **9**, 10402–10410.
- 30 H. Yoo, R. Engelke, S. Carr, S. Fang, K. Zhang, P. Cazeaux, S. H. Sung, R. Hovden, A. W. Tsen, T. Taniguchi, K. Watanabe, G. C. Yi, M. Kim, M. Luskin, E. B. Tadmor, E. Kaxiras and P. Kim, *Nat. Mater.*, 2019, **18**, 448–453.
- 31 S. Zhu and H. T. Johnson, *Nanoscale*, 2018, **10**, 20689–20701.
- 32 H. C. Movva, B. Fallahazad, K. Kim, S. Larentis, T. Taniguchi, K. Watanabe, S. K. Banerjee and E. Tutuc, *Phys. Rev. Lett.*, 2017, **118**, 247701.
- 33 B. Fallahazad, H. C. Movva, K. Kim, S. Larentis, T. Taniguchi, K. Watanabe, S. K. Banerjee and E. Tutuc, *Phys. Rev. Lett.*, 2016, **116**, 086601.
- 34 Y. Zhang, M. M. Ugeda, C. Jin, S.-F. Shi, A. J. Bradley, A. Martín-Recio, H. Ryu, J. Kim, S. Tang and Y. Kim, *Nano Lett.*, 2016, **16**, 2485–2491.
- 35 R. Bistritzer and A. H. MacDonald, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 12233–12237.
- 36 K. Kim, A. DaSilva, S. Huang, B. Fallahazad, S. Larentis, T. Taniguchi, K. Watanabe, B. J. LeRoy, A. H. MacDonald and E. Tutuc, *Proc. Natl. Acad. Sci. U. S. A.*, 2017, **114**, 3364–3369.
- 37 S. Xu, Z. Wu, H. Lu, Y. Han, G. Long, X. Chen, T. Han, W. Ye, Y. Wu and J. Lin, *2D Mater.*, 2016, **3**, 021007.
- 38 T. Han, J. Shen, N. F. Yuan, J. Lin, Z. Wu, Y. Wu, S. Xu, L. An, G. Long and Y. Wang, *Phys. Rev. B*, 2018, **97**, 060505.
- 39 M. H. Naik and M. Jain, *Phys. Rev. Lett.*, 2018, **121**, 266401.