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Furan and benzochalcogenodiazole based multichromic polymers *via* a donor–acceptor approach

Furan and benzochalcogenodiazole based multichromic polymers *via* a donor–acceptor approach†Cite this: *Polym. Chem.*, 2013, **4**, 2457Merve İçli-Özkut,<sup>a</sup> Halil İpek,<sup>b</sup> Baris Karabay,<sup>c</sup> Atilla Cihaner<sup>\*c</sup> and Ahmet M. Önal<sup>\*b</sup>

Two new furan and benzochalcogenodiazole based monomers, namely 4,7-di(furan-2-yl)benzo[c][1,2,5]-selenadiazole (**FSeF**) and 4,7-di(furan-2-yl)benzo[c][1,2,5]thiadiazole (**FSF**), were designed and synthesized *via* a donor–acceptor–donor approach. The monomers were electrochemically polymerized *via* potentiodynamic or potentiostatic methods. The monomers and their polymers exhibited lower oxidation potentials (1.16 V and 1.06 V for monomers; 0.93 V and 0.80 V for polymers vs. Ag/AgCl) and red shifts of the whole dual-band absorption spectra upon moving from S to Se. Intramolecular charge transfer properties of the monomers and the polymers were demonstrated by using electroanalytical and optical methods. Also, the polymers **PFSF** and **PFSF** were multicolored at different redox states and have low band gaps of 1.43 eV and 1.61 eV, respectively.

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## Introduction

After the investigation of the precursor conjugated polymer polyacetylene opened a new area of conducting polymers in the 1970s,<sup>1,2</sup> conjugated polyaromatics have been investigated extensively. Among them, myriad of studies have been reported on five-membered heterocycles containing polymers such as polypyrroles, polythiophenes and their derivatives<sup>3</sup> due to their susceptible electronic and optical properties. Very recently, many studies also have been performed on selenophene based polymers.<sup>4–10</sup> Among these studies, unfortunately the studies that contain furan and its derivatives have remained outnumbered due to the susceptibility of the furans on the side reactions during the polymerization and low stability of the resulting polymers.<sup>11–14</sup> In order to overcome these disadvantages, many methods have been tried including the use of boron trifluoride-diethyl ether to reduce the aromaticity of the furan ring to avoid overoxidation<sup>15</sup> and making the copolymers of furan with different monomeric units such as thiophene.<sup>16</sup> However, the corresponding polymers were not fully conjugated. The pursuit of lowering the oxidation potential of furan, *i.e.*, obtaining fully conjugated polyfurans, hybrid conjugated monomers have been started to be synthesized. Among them donor (D, electron-rich) and acceptor (A, electron-deficient) type monomers give

intriguing results<sup>3,17</sup> since it is possible to tune the band gap of the materials, which determines the optical and electronic properties, by the combination of D and A units in one pot. Generally, by using this approach, the highest occupied molecular orbital (HOMO) energy level of the D unit increases and the lowest unoccupied molecular orbital (LUMO) energy level of the A unit decreases and therefore, the gap between HOMO and LUMO levels decreases in a new hybrid monomer. If D and A units are combined under the same umbrella, an intramolecular charge transfer is generally observed which might affect the optical and electronic properties of their corresponding  $\pi$ -conjugated polymers.<sup>18,19</sup> The degree of charge density difference determines the charge transfer between the D and A units, which can be easily monitored using spectroscopic analysis.<sup>20</sup>

Recent studies showed that furan containing polymers have some priorities over thiophene based ones for the application of polymer organics in advanced technological applications. For instance, the greater electron withdrawing ability of furan is capable of reducing the HOMO energy level of the D parts in the solar cells which results in the high open circuit voltage.<sup>17</sup> Furthermore, the field-effect mobilities of furan derivatives are very similar to the thiophene based one, and this property together with natural feed-stock and higher open circuit voltage when compared its five-membered analogues make furan based polymers outstanding candidates for organic electronics.<sup>21</sup> There are also some examples of furan containing polymers, emitting blue light, obtained *via* polycondensation reactions.<sup>22,23</sup>

Keeping all this in mind, we have synthesized and characterized two new furan and benzochalcogenodiazole based monomers, namely 4,7-di(furan-2-yl)benzo[c][1,2,5]selenadiazole (**FSeF**) and 4,7-di(furan-2-yl)benzo[c][1,2,5]thiadiazole (**FSF**), *via* D–A–D approach. Then, the corresponding conjugated polymers, namely poly(4,7-di(furan-2-yl)benzo[c][1,2,5]selenadiazole) (**PFSF**) and

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**Scheme 1** Chemical structures of **FSeF**, **FSF** monomers and their corresponding polymers **PFSeF** and **PFSF**.

poly(4,7-di(furan-2-yl)benzo[*c*][1,2,5]thiadiazole) (**PFSF**), were electrochemically synthesized (Scheme 1) and also their optical and electrochemical properties were investigated. In addition, the intramolecular charge transfer properties of the monomers and their polymers were demonstrated by using electroanalytical and optical methods.

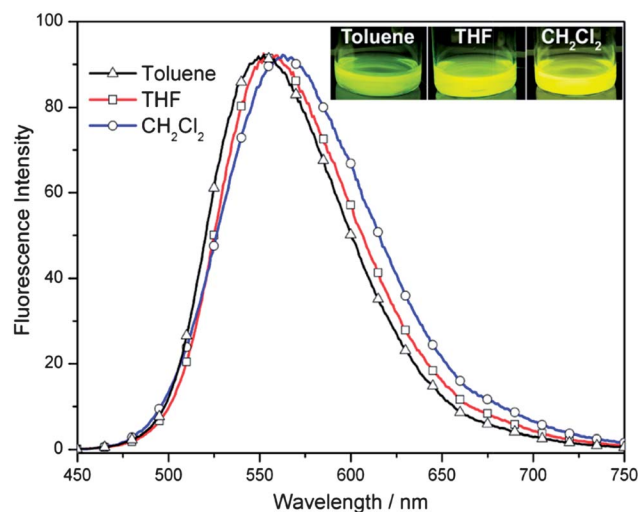
## Results and discussion

Two novel monomers, **FSeF** and **FSF**, having the same D units as furan and different A units of benzoselenadiazole and benzo-thiadiazole, were synthesized *via* Stille coupling reaction. Their characterizations were done by  $^1\text{H}$ ,  $^{13}\text{C}$  NMR, FTIR and mass measurements (see ESI $^\dagger$ ).

First of all, in order to understand the structure–property relationships in D–A–D type monomers, **FSeF** and **FSF**, where a single atom was changed in the A unit, the optical and electrochemical properties were investigated. As shown in Fig. 1a, the monomers exhibit dual-band absorptions due to the charge transfer between the D and A units since this is a characteristic signature generally observed in D–A type monomers. When

compared to **FSF**, the entire dual-band spectrum of **FSeF** is shifted to a lower energy due to the heavier chalcogen atom Se into the A unit. In the literature the absorption band at a longer wavelength was attributed to the intramolecular charge transfer in D–A type materials.<sup>20,24,25</sup> The low energy band at 481 nm for **FSF** was shifted to a longer wavelength at 481 nm for **FSeF** with decreasing band intensity, which may be due to the poor acceptor nature of the benzoselenadiazole unit containing the less electronegative Se atom since it is hardly be able to separate the charge between the D and A units (Fig. 1a).<sup>20,26</sup> Therefore, there is a poor intramolecular charge transfer in **FSeF**, which results in a red-shift absorption band with a low intensity. The emission colors of the monomers changed from greenish yellow to orange when moving from S to Se (Fig. 1b).

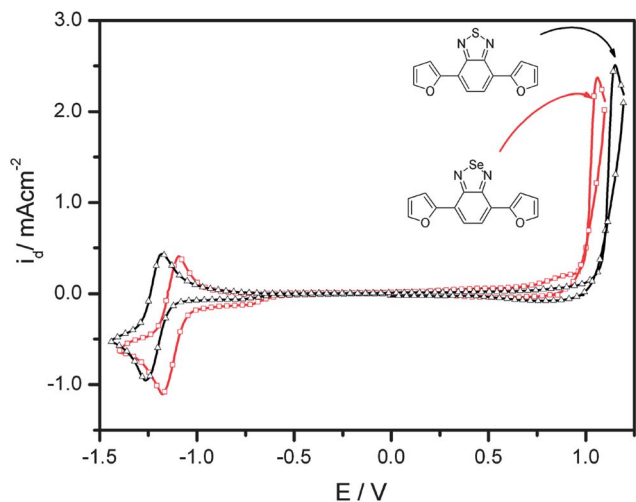
In order to support the intramolecular charge transfer between the D and A units, solvatochromic experiments were performed.<sup>20</sup> By changing the polarity of the solvents, the



**Fig. 2** Emission spectra of **FSF** in toluene (greenish yellow), THF (yellow) and  $\text{CH}_2\text{Cl}_2$  (yellow).



**Fig. 1** (a) Absorption and (b) emission spectra of **FSF** and **FSeF** in toluene. Inset: the colors of **FSF** and **FSeF** in toluene under (a) day light and (b) handheld UV lamp.



**Fig. 3** Cyclic voltammograms of  $3.3 \times 10^{-3}$  M of **FSeF** and  $4.1 \times 10^{-3}$  M of **FSF** in 0.1 M TBAH/ACN electrolyte solution at a scan rate of  $100 \text{ mV s}^{-1}$  vs. Ag/AgCl.

stability of the excited state must be checked and if there is enough charge separation in the monomer, the emission spectrum must shift to a longer wavelength (a red-shift) by increasing the polarity of the solvents. The emission spectra of **FSF** were tested in solvents of increasing dielectric constants: toluene (2.38), THF (7.58) and  $\text{CH}_2\text{Cl}_2$  (8.93).<sup>27</sup> As shown in Fig. 2, the emission spectra were shifted to longer wavelengths from 552.2 nm (toluene, greenish yellow) to 556.9 nm (THF, yellow) and to 562.7 nm ( $\text{CH}_2\text{Cl}_2$ , yellow). This red shift in the emission proves that the low-energy band present in the absorption spectrum is originating from intramolecular charge transfer.

On the other hand, the emission spectra of **FSeF** did not show any shift or change by increasing the solvent polarity, indicating the poor intramolecular charge transfer between furan and benzoselenadiazole units due to the poor electron acceptor power of the A unit (see ESI, Fig. S1†).<sup>20,26</sup>

After optical properties, the electrochemical behaviours of the monomers were investigated in an electrolyte solution consisting of 0.1 M TBAH dissolved in ACN. The **FSeF** and **FSF** monomers showed single irreversible oxidation peaks around

1.06 V and 1.16 V (vs. Ag/AgCl), respectively (Fig. 3). This difference was expected since the acceptor power of the benzothiadiazole is larger than the benzoselenadiazole due to the higher electronegativity of the S atom when compared to Se. Therefore, **FSeF** has higher electron rich nature than **FSF**, resulting in a lower oxidation potential. On the other hand, the benzotriazole containing monomer with the same D unit (2-dodecyl-4,7-di(furan-2-yl)-2H-benzo[d][1,2,3]triazole, **FNT**) synthesized in the literature has an irreversible oxidation potential at 1.04 V (after correcting for Ag/AgCl) due to the poor electron deficient nature of the A unit when compared to the benzoselenadiazole.<sup>28</sup> As a consequence, the oxidation potential values of the monomers depend on the electron deficient nature of the acceptor units by increasing order of N, Se and S. Unfortunately, there is no systematic relationship between redox behaviour and the A moiety of the monomers. The reduction behaviour of the monomers, for example, supported this observation since reduction potential values were opposite to the expected values. For example, in the same D–A–D system, if **FSeF** has a lower oxidation potential than **FSF**, its reduction potential must have a more negative value, but the monomers, **FSeF** and **FSF**, exhibited reversible reduction peaks with half wave potentials of  $E_{\text{m},1/2}^{\text{red}} = -1.13 \text{ V}$  and  $E_{\text{m},1/2}^{\text{red}} = -1.23 \text{ V}$ , respectively. In other words, **FSeF** is more easily reduced. The results are consistent with the data obtained in the literature with 3,4-propyldioxythiophene based (as D group) analogues of these polymers.<sup>29</sup> As a result, this difference observed in the cathodic region may not only be explained by the powers of the acceptor units but also depends on the factors such as heavy atom substitution, electronegativity, bond length and acceptor aromaticity.<sup>20</sup>

After determining the electrochemical behavior of **FSeF** and **FSF**, electrochemical polymerizations were carried out in ACN containing 0.1 M TBAH. During the repetitive scanning, new reversible redox couples were observed, which is a characteristic behavior of conducting polymer films, showing the formation of an electroactive polymer film on the electrode surface. Moreover, after each successive cycle the peak current values of the redox couples intensified, which confirmed an increase in the polymer film thicknesses (Fig. 4). The polymer films, **PFSeF** and **PFSF**, coated on the electrode surface were also scanned at



**Fig. 4** Electropolymerization of (a)  $2.3 \times 10^{-3}$  M of **FSeF** and (b)  $3.1 \times 10^{-3}$  M of **FSF** in 0.1 M TBAH/ACN at a scan rate of  $100 \text{ mV s}^{-1}$  by potential scanning to give **PFSeF** and **PFSF**, respectively.





**Fig. 5** Cyclic voltammograms of (a) **PFSeF** and (b) **PFSF** in 0.1 M TBAH/ACN electrolyte solution at a scan rate of 100 mV s<sup>-1</sup> vs. Ag/AgCl.

different scan rates. The linear increase as a function of the scan rates proved that the redox processes were non-diffusional controlled and also the polymer films were well-adhered on the electrode surface for both polymers (see ESI, Fig. S2 and S3†).

In endeavours to prove that there is n-type doping as well as the p-type one, the polymer films were scanned cathodically and anodically. When compared to S containing polymer **PFSF**, the n-type doping process of Se containing polymer **PFSeF** was easily observed (Fig. 5), which is consistent with the literature.<sup>29</sup>

During the anodic scan, reversible oxidation peaks were observed with half wave potentials of  $E_{p,1/2}^{ox} = 0.80$  V for **PFSeF** and  $E_{p,1/2}^{ox} = 0.93$  V for **PFSF**. These data were compatible with

those obtained from the monomers. Their benzotriazole based polymer analogue, namely (poly(2-dodecyl-4,7-di(furan-2-yl)-2H-benzo[d][1,2,3]triazole), **PFNT**) has a half wave potential at  $E_{p,1/2}^{ox} = 0.61$  V vs. Ag/AgCl.<sup>28</sup> For the sake of comparison, the potential values of the p-type doping processes agree well with the electron acceptor power of the A unit increasing upon moving from N to Se to S.

During the cathodic scan, polymers, **PFSeF** and **PFSF** showed half wave potentials of  $E_{p,1/2}^{red} = -1.16$  V and  $E_{p,1/2}^{red} = -1.24$  V, respectively. n-type doping of **PFNF** was reported as  $E_{p,1/2}^{red} = -1.79$  V vs. Ag/AgCl.<sup>28</sup> The potentials of the n-type doping processes shifted to more negative potentials upon changing Se

**Table 1** HOMO, LUMO and  $E_g$  values (from different measurements methods) of **PFSeF**, **PFSF** and **PFNF**

Monomers and polymers	HOMO <sup>a</sup> (eV)	LUMO <sup>a</sup> (eV)	$E_g^{CV}$ (V)	$E_g^{DPV}$ (V)	$E_g^{SPEL}$ (eV)
 FSeF	-5.52	-3.08	2.44	—	—
 FSF	-5.41	-3.31	2.10	—	—
 PFSeF	-4.90	-3.47	1.43	1.43	1.54
 PFSF	-4.99	-3.30	1.69	1.61	1.65
 PFNF <sup>**</sup>	-5.24 <sup>b</sup>	-3.22 <sup>b</sup>	2.02 <sup>b</sup>	—	1.9 <sup>b</sup>

<sup>a</sup> Calculated according to the literature: ref. 32 and 33. <sup>b</sup> Ref. 28.



Fig. 6 Absorption spectra of neutral state **PFSeF** and **PFSeF** films on ITO in ACN.

to S to N. In this sequence Se was expected to be between S and N atoms due to the power of the acceptor units, whereas this difference can be explained by the stabilization of the LUMO. As explained by Das *et al.* the degree of bond length alternation of S–N and Se–N in the A units increases on moving from S to Se, which decreases the aromaticity and leads to stabilization of the LUMO.<sup>20,26</sup> The HOMO, LUMO levels and  $E_g$  values were calculated from the cyclic voltammetry (CV) and are tabulated in Table 1 together with HOMO, LUMO levels of their corresponding monomers. As it is seen from Table 1 the HOMO energy levels of polymers increase in the order of N, S and Se.

It is interesting to note that, while thiophene analogues have p- and n-type doping similar to **FSeF** and **PFSeF**, pyrrole analogues have only p-type doping behaviour, which may be explained as due to the electron rich nature of the pyrrole unit when compared to furan.<sup>29–31</sup>

Another method to investigate the redox properties of the polymers is differential pulse voltammetry (DPV) method (see ESI, Fig. S4†). Band gaps calculated from DPV are in agreement with those calculated from CV (Table 1). On the other hand, the optical band gaps of the polymer films **PFSeF** and **PFSeF** were calculated as 1.54 eV and 1.65 eV from the onset of the low energy band, respectively, shown in Fig. 6.

These values are relatively bigger than their thiophene (1.46–1.50 eV) and pyrrole analogues (1.08–1.12 eV), which shows that optical spectra of the polymers red shift according to increasing power of donor unit.<sup>30,31</sup> The results are again consistent with the band gaps calculated from CV (1.43 eV for **PFSeF** and 1.69 eV for **PFSeF**) and DPV (1.43 eV for **PFSeF** and 1.61 eV for **PFSeF**).









The entire dual-band absorption spectrum of **PFSeF** shifts to lower energy and the absorption band at longer wavelength of the corresponding polymer is attributed to the intramolecular charge transfer as in the case of monomer. When compared to **PFSeF**, the low energy band of **PFSeF** has a red shift with lower intensity due to the presence of heavier atom (lower ionization energy of Se) and the poor acceptor nature of benzoselenadiazole. As a consequence, a band gap reduction was observed on moving from S to Se. On the other hand, **PFNF** film ( $E_g^{\text{SPEL}} = 1.9$  eV) exhibited only one broad absorption band centred at 524 nm,<sup>28</sup> which could be attributed to the strength of the acceptors, among which benzotriazole is the poorest A unit and there is no enough ability for the charge separation between D and A units in polymer backbone.<sup>26</sup> Similar behaviour was also observed in our previous work about 3,4-propylenedioxythiophene and benzotriazole based polymers.<sup>29</sup> On the other hand, the polymers containing benzothiadiazole and benzoselenadiazole units as acceptor and thiophene and pyrrole units as donor exhibit dual band spectra as reported previously.<sup>30,31</sup>

Upon oxidation the transition band of **PFSeF** at 355 nm started to decrease with a concomitant increase in the intensity of the band at 577 nm. Also, the formation of a new band beyond 700 nm was observed confirming the formation of the charge carriers in Fig. 7a. During the oxidation, the color of the **PFSeF** changed from dark gray (0.0 V) to dark cyan (0.2 V) and to blue (1.15 V). Furthermore, during p-type doping, an isosbestic point at 410 nm indicates that only two phases coexist. On the other hand, the transition band at 335 nm shifted to a longer wavelength and the band at 565 nm firstly began to increase and then decrease upon applied higher oxidation potentials. An increase in the intensity of new absorption band beyond 700 nm also represented charge carrier formation (Fig. 7b). Upon oxidation the color of the **PFSeF** turned from wine (0.0 V) to violet (0.2 V) and to dark gray (1.05 V). It should be also mentioned



Fig. 7 Electronic absorption spectra of the (a) **PFSeF** and (b) **PFSeF** on ITO in 0.1 M TBAH/ACN at various applied potentials along the anodic region.

**Table 2** Optical properties of the **PFSeF** and **PFSF**

Polymers	$\lambda_{\text{max},1}$ (nm)	$\lambda_{\text{max},2}$ (nm)	Colorimetric results				Colors at different oxidation states				
			0.0 V	0.2 V	1.15 V	−1.3 V	0.0 V	0.2 V	1.15 V	−1.3 V	
PFSeF	355	577	<i>L</i>	39.20	53.62	42.83	54.84				
			<i>a</i>	0.015	−0.20	4.99	−2.59				
			<i>b</i>	13.82	−17.46	−17.79	11.96				
Polymers	$\lambda_{\text{max},1}$ (nm)	$\lambda_{\text{max},2}$ (nm)	Colorimetric results				Colors at different oxidation states				
			0.0 V	0.2 V	1.05 V	−1.4 V	0.0 V	0.2 V	1.05 V	−1.4 V	
PFSF	335	565	<i>L</i>	30.02	51.88	47.55	59.48				
			<i>a</i>	11.59	16.89	7.71	0.83				
			<i>b</i>	9.39	−14.46	−9.43	−7.94				

that the color of the **PFSeF** and **PFSF** were dark yellow and light gray in the reduced states, respectively (Table 2 and see ESI, Fig. S5†).

Stability under ambient conditions is an important parameter for any material to be used in advanced technological applications. Also, the stability of the polymer films was carried out under atmospheric conditions (without purging and sweeping the electrolyte solution with inert gas) between their redox states by using cyclic voltammetry method (see ESI, Fig. S6†). It was observed that **PFSeF** retained 85% of its electroactivity whereas **PFSF** retained just 39%.

## Conclusions

Two new furan based D–A–D type monomers, **FSeF** and **FSF** and their corresponding polymers, **PFSeF** and **PFSF**, were synthesized and also their optical and electrochemical properties were investigated. The intramolecular charge transfer properties of the monomers and their polymers were demonstrated by using electroanalytical (CV and DPV) and optical (UV-vis and fluorescence) methods. It can be easily conjectured that the power of the acceptor moieties, heavy atom substitution, electronegativity, ionization energy, bond length alternation and acceptor aromaticity have important roles in electrochemical and optical studies. Also, the polymers were multicolored at different redox states and they have low band gaps (1.43 eV for **PFSeF** and 1.61 eV for **PFSF**).

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