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# Synthesis, growth mechanism and photocatalytic H<sub>2</sub> evolution of CdS/CuS composite *via* hydrothermal method

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In recent years, visible light-driven photocatalysts used for confronting energy shortages and environmental pollution have drawn much attention. CdS is regarded as an excellent photoelectric semiconductor for photocatalysis, but photocorrosion and low photocatalytic activity limit its practical application. In order to improve the photocatalytic performance of CdS, we synthesized a II-type CdS/CuS composite *via* a hydrothermal method in one step. CdS, CuS and the CdS/CuS composite have flower-like structures according to FESEM results. XRD and EDS results confirm that the composite is composed of CdS and CuS, indicating that we have successfully synthesized the CdS/CuS composite. UV-Vis and PL results show that the formation of heterojunction structures with CuS can be used to control the optical properties of CdS. H<sub>2</sub> evolution results show that the CdS/CuS composite generates H<sub>2</sub> at a rate of 295  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, which is higher than that of CdS.

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# 1. Introduction

Energy consumption and environmental pollution have become a crisis due to the development of modern living standards; it is a challenge for us to find an efficient way to solve these problems.<sup>1</sup> In past research reports, hydrogen energy has been considered a sustainable and clean form of energy, and it can be abundantly obtained by splitting of water into hydrogen using nanomaterials, which is a promising and attractive strategy for confronting the above issues.<sup>2-4</sup> Thus, various nanomaterials have been used as effective photocatalysts for splitting water to produce H<sub>2</sub>, such as TiO<sub>2</sub>,<sup>5</sup> ZnO,<sup>6</sup> g-C<sub>3</sub>N<sub>4</sub>,<sup>7</sup> CuS<sup>8</sup> and BiVO<sub>4</sub>.<sup>9</sup> In addition, photocatalytic H<sub>2</sub> evolution is critically related to the size, structure, surface area and morphology of the nanomaterials used.10 Therefore, various morphologies of nanomaterials have been reported, such as ZnO nanorods,<sup>11</sup> TiO<sub>2</sub> nanospheres,12 MoS<sub>2</sub> nanosheets,13 and flower-like CdS structures.<sup>14</sup> However, in the field of photocatalytic nanomaterials, the major challenge is to find efficient and stable nanomaterials for photocatalytic application under solar light.

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Among these types of nanomaterials, CdS has attracted much attention due to its conduction band position and band gaps being suitable for energy conversion under visible light.15 As is known, CdS (2.42 eV) is an important visible light photocatalyst because of its sensitivity to visible light irradiation and efficient photoexciton generation, which are excellent qualities for a photocatalyst.<sup>16,17</sup> However, CdS has some disadvantages for further application. First, CdS nanostructures can easily aggregate during photocatalytic reactions, which can reduce the surface area. Second, the photoexciton recombination is ultrafast, and there is a lack of reactive sites.18 Third, instability occurs due to photocorrosion or photodissolution.19 Therefore, CdS usually exhibits low photocatalytic activity due to the above drawbacks. To solve these problems, significant efforts have been devoted to improving the photocatalytic activity of CdS. Examples include the synthesis of CdS quantum dots with high surface areas,20 deposition of noble metals,21 loading cocatalysts onto the CdS surface,<sup>22</sup> metal ion doping,<sup>23</sup> and the incorporation with other semiconductors to form heterojunction structures.24

In comparison with noble metal cocatalyst addition and metal ion doping, the hybridization of CdS with other semiconductors having matching band gaps to form heterojunction structures is beneficial for suppressing the recombination of photoinduced charge carriers. Furthermore, this approach can effectively improve spatial charge separation and prolong the lifetime of the charge carriers, which can enhance the photocatalytic properties.<sup>25</sup> In addition, this charge separation is better facilitated by a II-type system than a I-type system.<sup>26</sup> Therefore, we tried to synthesize a II-type CdS heterojunction

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#### Paper

structure in one step to enhance the photocatalytic activity. CuS is an important semiconductor due to its excellent physical and chemical properties with a band gap of about 2.0 eV, with potential applications in photocatalysis, photothermal conversion, and solar cells.<sup>27</sup> Markovskaya et al. prepared the noblemetal-free photocatalyst CuS/Cd<sub>0.3</sub>Zn<sub>0.7</sub>S and found that the photocatalytic activity reached as high as 3520 µmol g<sup>-1</sup> h<sup>-1</sup>.28 Markovskaya et al. also synthesized Cd<sub>0.3</sub>Zn<sub>0.7</sub>S nanoparticles with  $Cu_x S$  and  $Cu^0$  as co-catalysts, which achieved a photocatalytic hydrogen evolution of 6.4 mmol g<sup>-1</sup> h<sup>-1</sup>.<sup>29</sup> Chen et al. presented well-defined Cu<sub>1.94</sub>S-Zn<sub>x</sub>Cd<sub>1-x</sub>S heteronanorods for photocatalytic hydrogen evolution, which showed a hydrogen production activity of 7735  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>.<sup>30</sup> Benefiting from the excellent optical performance and matched band gap of CuS, we fabricated a II-type CdS/CuS composite through a hydrothermal method in one step. We have described the growth mechanism, optical properties and H2 evolution reaction of CdS/CuS composites with different molar ratios of Cd : Cu. Compared with pure CdS, the CdS/CuS composite exhibited better photocatalytic efficiency for H<sub>2</sub> evolution and generated H<sub>2</sub> at a rate of 295  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> when the molar ratio of Cd : Cu was 0.5 : 0.5. Thus, CuS plays an important role in improving the photocatalytic activity.

## 2. Experimental

#### 2.1 Synthesis of CdS, CuS, and CdS/CuS composite

All the chemicals used in this work were of analytical reagent grade. Cadmium nitrate  $(Cd(NO_3)_2 \cdot 4H_2O)$ , cupric nitrate  $(Cu(NO_3)_2 \cdot 3H_2O)$ , and thiourea were used without any further purification. The aqueous solutions were made using deionized water.

Scheme 1: first, 0.008 mol  $Cd(NO_3)_2 \cdot 4H_2O$  and 0.002 mol  $Cu(NO_3)_2 \cdot 3H_2O$  were added into a beaker containing 90 mL deionized water with stirring for 10 minutes. Then, we added 0.032 mol thiourea into the aqueous solution (Cd : S = 1 : 3, Cu : S = 1 : 4, molar ratio). After stirring for 10 minutes, the aforementioned solution was transferred to a Teflon-lined autoclave of 120 mL capacity.

Scheme 2: first, 0.002 mol Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O and 0.02 mol thiourea were added into a beaker containing 90 mL deionized water with stirring for 10 minutes. Then, we added 0.008 mol Cd(NO<sub>3</sub>)<sub>2</sub>·4H<sub>2</sub>O into the aqueous solution. After stirring for 10 minutes, the aforementioned solution was transferred to a Teflon-lined autoclave of 120 mL capacity. The amount of thiourea was fixed at 0.02 mol.

Scheme 3: first, 0.008 mol  $Cd(NO_3)_2 \cdot 4H_2O$  and 0.02 mol thiourea were added into a beaker containing 90 mL deionized water with stirring for 10 minutes. Then, we added 0.002 mol  $Cu(NO_3)_2 \cdot 3H_2O$  into the aqueous solution. After stirring for 10 minutes, the aforementioned solution was transferred into a Teflon-lined autoclave of 120 mL capacity. The amount of thiourea was again fixed at 0.02 mol.

All the autoclaves were sealed and maintained at 160 °C for 24 h. Afterwards, the autoclaves were cooled to room temperature naturally. The resulting products were repeatedly washed with deionized water and absolute ethanol several times. Finally, the final products were dried at 60 °C in air. We varied the concentration of cupric nitrate ( $Cd^{2+}$  :  $Cu^{2+} = 0.6 : 0.4$ , 0.5 : 0.5, 0.4 : 0.6, 0.2 : 0.8, molar ratio) to synthesize the CdS/ CuS composites. For the purpose of comparison, the pure CdS and CuS were prepared under the same conditions.

#### 2.2 Characterization

The crystal phases of CdS, CuS, and CdS/CuS were measured by powder X-ray diffraction (PXRD, D8-Focus, Bruker AXS) with Nifiltered and Cu K $\alpha$  radiation ( $\lambda = 1.5406$  Å). The morphologies of the as-prepared samples were characterized by field emission scanning electron microscopy (FESEM, SU8010, HITACHI). A scanning electron microscope (SEM, Coxem EM-30AX PLUS<sup>+</sup>) equipped with an energy dispersive spectrometer (EDS, Bruker Nano Xflash610-H) was used to analyze the chemical composition at the sample surface. The optical absorption and room temperature photoluminescence spectra (PL) of the samples were obtained on an ultraviolet-visible spectrophotometer (UV-Vis, UV-2600, Shimadzu) and a fluorescence spectrophotometer (F-4500, HITACHI) equipped with a Xe lamp (excitation wavelength is 338 nm).

#### 2.3 Photocatalytic H<sub>2</sub> evolution

Photocatalytic  $H_2$  evolution was performed in a quartz reactor under visible light irradiation. 0.1 g sample was dispersed in 90 mL DI water in a quartz reactor, and then 10 mL triethanolamine and 1 mL  $H_2$ PtCl<sub>6</sub> aqueous solution were added into the aqueous solution under stirring. In this process, triethanolamine was added as a sacrificial electron donor, and  $H_2$ PtCl<sub>6</sub> was added to reduce Pt as a cocatalyst. After that, the quartz reactor was purged with  $N_2$  for 15 min to remove air and sealed with a rubber stopper. The Pt cocatalyst was reduced by  $H_2$ PtCl<sub>6</sub> under irradiation for 1 h with visible light. Finally, the photocatalytic water splitting reaction was irradiated under the same visible light. The  $H_2$  content was analyzed by a gas chromatograph at a rate of 1 mL gas each hour.

### 3. Results and discussion

#### 3.1 FESEM results of CdS, CuS and CdS/CuS composite

Fig. 1 shows the FESEM images of CdS and CuS. We can see from Fig. 1 that the morphology of CdS is a flower-like structure, assembled by nanoparticles. The diameter of the flower-like CdS is about 5  $\mu$ m. The morphology of CuS is also a flower-like structure assembled by nanosheets, with a diameter of about 2  $\mu$ m. This means that both CdS and CuS are flower-like structures.

Fig. 2 shows the FESEM images of the CdS/CuS composite, synthesized through Scheme 1, at different molar ratios of Cd : Cu. As shown in Fig. 2, the CdS/CuS composites are flower-like structures with different molar ratios of Cd : Cu. The nanoparticles are composed of CdS, and the nanosheets are composed of CuS. When the Cu source is increased, the flower-like structure becomes increasingly assembled by nanosheets.

Fig. 3 shows the FESEM images of the CdS/CuS composite, synthesized through Scheme 2. Fig. 3 shows that the flower-like



#### Fig. 1 FESEM images of CdS and CuS.

CdS/CuS composites are mainly assembled by nanosheets with different molar ratios of Cd : Cu using Scheme 2. The amount of the Cu source did not obviously affect the morphologies of the CdS/CuS composites as the molar ratios of Cd : Cu increased.

Fig. 4 shows the FESEM images of the CdS/CuS composite, synthesized through Scheme 3. We can see from Fig. 4 that the morphologies of the CdS/CuS composites are mainly assembled by nanoparticles at high molar ratios of the Cd source. However, when the amount of the Cu source is increased, the flower-like structure becomes increasingly assembled by nanosheets.

In Scheme 1, a Cd source and Cu source were combined in a beaker, with an amount of thiourea that was sufficient for  $Cd^{2+}$  and  $Cu^{2+}$  to form a strong ligand compound. The CdS/ CuS composites were mainly assembled by nanoparticles and nanosheets with low amounts of the Cu source, and the composites were mainly assembled by nanosheets with high amounts of the Cu source. In Scheme 2, the Cu source and thiourea were first added to the beaker; thus,  $Cu^{2+}$  may firstly form a strong ligand compound with thiourea. As the stability of Cu ligand compound is stronger than that of the Cd ligand compound, the CdS/CuS composites were mainly assembled by nanosheets with limited amounts of thiourea at different molar ratios of Cd : Cu. In Scheme 3, the Cd source and thiourea were first added into the beaker; thus,  $Cd^{2+}$  may firstly form a ligand compound with thiourea. The CdS/CuS composites were mainly assembled by nanoparticles with high molar ratios of the Cd source. According to the FESEM results shown in Fig. 2 to Fig. 4, we chose the as-prepared samples from Scheme 3 for characterization by XRD, EDS, UV-Vis, PL and H<sub>2</sub> evolution analyses.

#### 3.2 XRD results of CdS, CuS and CdS/CuS composite

Fig. 5 displays the XRD patterns of CdS, CuS and the CdS/CuS composite at different molar ratios of Cd : Cu before and after photocatalytic H<sub>2</sub> evolution. Fig. 5A shows that the diffraction peaks of CdS and CuS were well-indexed to the hexagonal phase of CdS (JSPDS card no. 41-1049) and the hexagonal phase of CuS (JSPDS card no. 06-0464). No peaks belonging to any other phases or impurities were detected. When the molar ratio of Cd : Cu was 0.8 : 0.2, the mixed peaks were attributed to CdS and CuS. The diffraction peaks belonged to CuS when  $2\theta$  was 29.2°, 31.7°, and 32.8°. When the molar ratio of Cd : Cu was 0.6 : 0.4, we can find the peak attributed to CuS at  $2\theta = 59.2^{\circ}$ . When the molar ratio of Cd : Cu was 0.5 : 0.5, we can find the peak belonging to CuS at  $2\theta = 10.7^{\circ}$ . Furthermore, the



Fig. 2 FESEM images of CdS/CuS composite at different ratios of Cd : Cu through Scheme 1.



Fig. 3 FESEM images of CdS/CuS composite at different ratios of Cd : Cu through Scheme 2.

intensity of the diffraction peaks belonging to CuS increased when the molar ratios of Cd : Cu changed from 08 : 0.2 to 0.2 : 0.8. The diffraction peak at  $2\theta = 47.9^{\circ}$  belongs to CdS and CuS, and the intensity of this diffraction peak also increased with the change in the molar ratios of Cd : Cu. On the other hand, the intensity of some diffraction peaks belonging to CdS decreased at different molar ratios of Cd : Cu. Therefore, we confirmed that we successfully synthesized the CdS/CuS composite based on the XRD results. In addition, the samples exhibit stable photocatalytic activity for H<sub>2</sub> evolution. Fig. 5B shows a small difference in the XRD patterns after the H<sub>2</sub> evolution test, which implies that the CdS, CuS and CdS/ CuS samples exhibit excellent stability for H<sub>2</sub> evolution under visible light irradiation.

#### 3.3 EDS results of CdS, CuS and CdS/CuS composite

Fig. 6 presents the EDS results for CdS, CuS and the CdS/CuS composite when the molar ratio of Cd : Cu is 0.5 : 0.5, synthesized following Scheme 3. We can see from Fig. 6A that the elemental composition calculated using EDS analysis was 23 and 77 wt% of S and Cd, respectively, and the atomic ratio of S : Cd was about 1:1. Shown in Fig. 6B, the elemental composition calculated by EDS was 36 and 64 wt% of S and Cu, respectively, with an atomic ratio of S : Cu that was about 1:1. These results show that the samples contained pure CdS and CuS. As shown in Fig. 6C, the elemental composition was 28, 28, 44 wt% of S, Cu, Cd, respectively, calculated using EDS analysis, with the atomic ratio of S : Cu of about 1 : 1. Therefore, the composites are composed of CdS and CuS. Fig. 7 shows the EDS mapping results of the CdS/CuS composite when the molar



Fig. 4 FESEM images of CdS/CuS composite at different ratios of Cd : Cu through Scheme 3.



Fig. 5 XRD patterns of CdS, CuS and CdS/CuS composite at different ratios of Cd : Cu. (A) Before photocatalysis. (B) After photocatalysis.

ratio of Cd : Cu was 0.5 : 0.5, synthesized following Scheme 3. We can see from Fig. 7 that the elemental composition of the CdS/CuS composite includes Cd, Cu, and S elements. The nanoparticles belong to the Cd element and the nanosheets belong to the Cu element. Therefore, we confirm that we have successfully synthesized the CdS/CuS composite from EDS results.

#### 3.4 UV-Vis and PL tests of CdS, CuS and CdS/CuS composite

Fig. 8 presents the UV-Vis absorption spectra and PL spectra for CdS, CuS and the CdS/CuS composite at different ratios of Cd : Cu. It can be seen from Fig. 8A that the flower-like CdS structure strongly absorbs in the visible light region upto

520 nm. The flower-like CuS structure showed a strong absorption peak at 620 nm. When CdS and CuS form the CdS/ CuS composite, the absorption edge shifts toward longer wavelengths because of the contribution from CuS, which has an absorption peak at 620 nm. Thus, the absorption edge of the CdS/CuS composite shifts toward a wavelength of about 600 nm when the amount of Cu source is increased. Therefore, forming the heterojunction structure with CuS can be used to improve the absorption capability of CdS for visible light. Fig. 8B shows the PL spectra of CdS, CuS and the CdS/CuS composite at different molar ratios. We can see from Fig. 8B that all the samples have a similar emission peak at 370 nm, and the flower-like CdS has another wide emission peak at 582 nm. With the



Fig. 6 EDS analysis of CdS, CuS and CdS/CuS composite.

Paper



Fig. 7 EDS mapping of CdS/CuS composite when the molar ratio of Cd : Cu is 0.5 : 0.5.

increase in the amount of Cu source, the peak near 582 nm disappears. The possible cause is that the composites have a CuS composition. As we know, PL spectra reveal the efficiency of charge carrier trapping, immigration and transfer.<sup>31</sup> Thus, the decreased intensity of the PL spectra indicate the lower

recombination rate of the photogenerated electrons and holes, leading to higher photocatalytic activity. Therefore, we can confirm based on Fig. 8 that the formation of a heterojunction structure with CuS can be used to control the optical properties of CdS.



Fig. 8 (A) UV-Vis absorption spectra, (B) PL spectra of CdS, CuS and CdS/CuS composite at different ratios of Cd : Cu.



#### Growth mechanism of CdS, CuS and CdS/CuS composite 3.5

Fig. 9 shows the growth mechanism of CdS, CuS and the CdS/ CuS composite. For CdS, Cd<sup>2+</sup> and thiourea formed a strong Cd-thiourea ligand compound;  $Cd^{2+}$  and  $S^{2-}$  were then slowly released and formed CdS nanoparticles. Additionally, the anisotropy of CdS would increase. Several CdS nanoparticles assembled together to form flower-like structures.32 For CuS, Cu<sup>2+</sup> and thiourea formed a Cu-thiourea ligand compound, and then, the ligand compound released Cu<sup>2+</sup> and S<sup>2-</sup> slowly and formed CuS nanosheets upon heating. Then, several CuS nanosheets assembled together to form flower-like structures. For the CdS/CuS composite, Cd2+ and Cu2+ each formed a strong ligand compound with thiourea. After heating at a high temperature, the ligand compounds slowly released Cu<sup>2+</sup>, Cd<sup>2+</sup> and S<sup>2-</sup>, which formed CdS nanoparticles and CuS nanosheets.<sup>33</sup> On account of the limited thiourea, Cu<sup>2+</sup> and Cd<sup>2+</sup>

share the thiourea, which releases S<sup>2–</sup>. Then, CdS nanoparticles and CuS nanosheets assemble to form a flower-like composite. Therefore, we can fabricate flower-like CdS/CuS composites in one step through a hydrothermal method.

#### 3.6 Photocatalytic H<sub>2</sub> evolution of CdS, CuS and CdS/CuS composite

The visible light photocatalytic H<sub>2</sub> evolution reaction (HER) of CdS, CuS and the CdS/CuS composite is shown in Fig. 10A. The results show that all the samples perform H<sub>2</sub> evolution because CdS and CuS have absorption in the visible light region. We can see from Fig. 10A that pure CdS and pure CuS each generate H<sub>2</sub> at rates of 207  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> and 219  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, respectively. After we synthesized the CdS/CuS composites with different ratios of Cd : Cu, the increased efficiency of the H<sub>2</sub> production compared to that of pure CdS or pure CuS becomes evident.



Fig. 10 (A) Time courses of H<sub>2</sub> evolution with CdS, CuS and CdS/CuS composite, (B) schematic diagram of electronic transfer mechanism.

When the ratio of Cd : Cu is 0.5 : 0.5, the CdS/CuS composites generate  $H_2$  at a rate of 295 µmol g<sup>-1</sup> h<sup>-1</sup>, which benefits from the constructed CdS/CuS heterojunction. The increase in the size of the CdS/CuS composite did not obviously increase the efficiency of H<sub>2</sub> production. These results show that the microsized CdS/CuS composite could be synthesized as nanosized to further enhance H<sub>2</sub> production because of nanosized materials have larger specific surface. As is shown in Fig. 10B, when CuS combines with CdS to construct a II-type heterojunction, the visible light-excited electrons in the conduction band (CB) of CuS can transfer into the CB of CdS, and the holes in the valence band (VB) of CdS can transfer into the VB of CuS, which can enhance the separation of the photoinduced charge carriers and holes.<sup>34,35</sup> In this way, the efficient separation of photogenerated electron-hole pairs can be achieved, which further improves the visible light photocatalytic H<sub>2</sub> evolution activity. Therefore, forming heterojunction structure with CuS can be used to enhance H<sub>2</sub> production of CdS.

# 4. Conclusions

In summary, we have synthesized a CdS/CuS composite through a hydrothermal method in one step. FESEM images show a flower-like morphology for CdS, CuS and the CdS/CuS composite synthesized following different schemes. EDS results also confirm that all the composites are composed of CdS and CuS. XRD results show that pure CdS and pure CuS are in the hexagonal phase, and the CdS/CuS composites have diffraction peaks belonging to CdS and CuS with different ratios of Cd : Cu. UV-Vis and PL spectroscopy results show that forming a heterojunction structure with CuS can be used to control the optical properties of CdS. H<sub>2</sub> evolution results show that the CdS/CuS composite generates H<sub>2</sub> at a rate of 295 µmol  $g^{-1}$  h<sup>-1</sup>, which is higher than that of pure CdS. Therefore, forming a heterojunction structure with CuS can be used to enhance the H<sub>2</sub> production of CdS.

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# Conflicts of interest

There are no conflicts to declare.

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