

## PAPER

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# Effective surface passivation of GaAs nanowire photodetectors by a thin ZnO capping

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The III–V nanowire (NW) structure is a good candidate for developing photodetectors. However, high-density surface states caused by the large surface-to-volume ratio severely limit their performance, which is difficult to solve in conventional ways. Here, a robust surface passivation method, using a thin layer of ZnO capping, is developed for promoting NW photodetector performance. 11 cycles of ZnO, grown on pure zinc blende high-quality GaAs NWs by atomic layer deposition, significantly alleviates the undesirable effect of the surface states, without noticeable degradation in NW morphology. An average 20-fold increase in micro-photoluminescence intensity is observed for passivated NWs, which leads to the development of detectors with high responsivity, specific detectivity, and optical gain of  $9.46 \times 10^5 \text{ A W}^{-1}$ ,  $3.93 \times 10^{14}$  Jones, and  $2.2 \times 10^8 \%$ , respectively, under low-intensity 532 nm illumination. Passivated NW detectors outperform their counterparts treated by conventional methods, so far as we know, which shows the potential and effectiveness of thin ZnO surface passivation on NW devices.

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## 1. Introduction

Photodetectors based on semiconductor nanowires (NWs) with a quasi-one-dimensional morphology have received immense attention over recent years, owing to their dislocation-free growth on foreign substrates, strong light-matter interaction, and excellent light-trapping effect.<sup>1–3</sup> The possibility of high quantum efficiency and photoresponsivity motivates future research on nanowire photodetectors. Numerous methods are demonstrated to boost nanowire photodetector behavior, including doping,<sup>4</sup> passivation,<sup>5</sup> surface plasmon,<sup>6</sup> improvement in material quality,<sup>7</sup> and more complex configurations.<sup>8,9</sup> It is worth noting that III–V nanostructures attract widespread attention, because of their wide tunable bandgaps, high integration potential, and high absorption efficiency.<sup>10,11</sup>

However, the large surface-to-volume ratio of NWs relative to the thin-film structure causes a high density of the surface states that could act as efficient recombination and scattering centers for carriers. The majority of III–V materials suffer from a large surface recombination velocity due to the surface states, such as that for GaAs, which is estimated to be the order of  $10^6 \text{ cm s}^{-1}$ ,<sup>12</sup> which significantly impedes the development and optimization of optoelectrical devices. In particular, for small-diameter nanowires, the conductivity could even be switched off due to complete depletion by surface band bending,<sup>13</sup> rendering them unsuitable for some devices.

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*quality material growth and device fabrication for III–V semiconductor nanowire microelectronics and optoelectronics.*

Passivation is thus needed to produce a stable inert film to saturate dangling bonds and prevent surface oxidation. In general, common cost-effective surface passivation for III-V materials includes dielectric atomic layer deposition (ALD),<sup>14–16</sup> sulfur treatment,<sup>17</sup> nitridation,<sup>18</sup> a sol-gel SiO<sub>2</sub> shell,<sup>19</sup> annealing, acid, and plasma treatment.<sup>20</sup> However, chemical surface treatment, such as sulfur passivation, is not a preferable choice for long-term passivation as its effect could vanish easily under an ambient atmospheric environment.<sup>21</sup> The plasma treatment energy needs to be carefully optimized to achieve less roughness, due to ion bombardment. Although nitridation could guarantee surface state suppression for a time, the use of toxic and dangerous reagents is an important issue.

As an extensively researched wide bandgap semiconductor material, zinc oxide (ZnO) is focused on because of its properties of long-term stability, optical transparency, passivation effect, carrier-selective transport, and high electron mobility.<sup>22</sup> Some efforts have been made to combine ZnO and III-V materials for behavioral improvement, in fabricating high-performance MOSFETs, solar cells, photodetectors, or capacitors.<sup>23–26</sup> Nevertheless, the potential for modifying photodetectors based on III-V NWs has still not been fully explored up to date. Although there exist some reports based on a combination of III-V NWs and ZnO,<sup>27–30</sup> they commonly use thick n-type ZnO shells, which are unable to extract the sole role of passivation because of the formation of a possible strong pn junction and strain defects due to lattice mismatch. The use of an ultrathin oxide passivation layer can effectively solve all these side effects brought about by a thick passivation layer, as the thin thickness is not enough to accumulate a large enough strain to change the sample morphology or band bending. Previous reports have claimed that an ultrathin ZnO layer is sufficient for passivation.<sup>26,31,32</sup> However, an ultrathin oxide passivation layer has rarely been used on NW structures, especially for photodetector fabrication. Our work demonstrates that this ultrathin ZnO layer strategy by atomic layer deposition is a promising passivation method for the nanowires without causing degradation in material quality.

GaAs (with a moderate direct bandgap of 1.42 eV) is a valuable and competitive candidate for optical detecting functions as it shows good ability for light absorption and light conversion due to its direct bandgap. Previously, we have grown high-quality phase-pure GaAs NWs by a self-catalyzed method, which could avoid generating deep trap levels by incorporation of foreign catalyst impurities and have developed optical detectors, demonstrating outstanding responsivity and specific detectivity.<sup>33</sup> Based on that work, an efficient passivation scheme is explored for further improvements.

In this report, the development of NW surface treatment is continued using pure zinc blende (ZB) high-quality GaAs NWs that can prevent the side effects of stacking faults, such as carrier scattering and transportation barriers. A thin ZnO surface passivation layer of only 11 cycles is grown conformally by atomic layer deposition (ALD) on NWs to achieve persistent protection and avoid the side effect of using a thick layer. The

effective passivation of thin ZnO on these GaAs NWs is shown by optical properties and Schottky-type photodetector performance.

## 2. Experimental methods

### GaAs nanowire growth and characterization

GaAs NWs are grown by a self-catalyzed method on p-type Si substrates *via* solid-source molecular beam epitaxy (MBE) with nominal p-doping (beryllium) in the order of 10<sup>18</sup> cm<sup>-3</sup>. These nanowires have a pure ZB crystal structure up to the top-most bilayer, which is optional for developing superior photodetectors owing to the absence of scattering centers and carrier-transportation barriers caused by stacking faults. The nanowires used in this study come from the same sample as in our previous publication, where TEM images and electron diffraction patterns show a pure ZB crystal structure.<sup>33</sup> The average length and diameter of doped GaAs nanowires are measured to be 4 μm and 60 nm, respectively. For undoped GaAs nanowires, the average length and diameter are 10 μm and 180 nm, respectively.

The micro-photoluminescence (μ-PL) intensity spectra of at least 12 individual intrinsic GaAs nanowires were measured at room temperature with a 532 nm excitation laser. The laser spot size was about 1 μm in diameter, focused by an objective lens. The morphology of the fabricated device was observed by scanning electron microscope (SEM). The chemical states of ZnO were investigated *via* X-ray photoelectron spectroscopy (XPS).

### Device fabrication and measurements

To fabricate a single GaAs nanowire-based photodetector, a heavily p-doped silicon substrate with thermally grown 100 nm SiO<sub>2</sub> was utilized. As-grown GaAs nanowires were transferred onto an SiO<sub>2</sub>/Si substrate by ultrasonic treatment in isopropyl alcohol (IPA) solution and subsequent drop-casting. The GaAs nanowires were immersed in dilute hydrochloric acid (HCl) for about 60 seconds to remove their native oxide. Afterward, ZnO thin films were grown at 120 degrees by thermal ALD, which is superior for damage-free and conformal deposition. Diethylzinc (DEZn) and deionized water were used as precursors to grow the ZnO thin films with a nominal growth rate of 0.17 nm per cycle. 11 cycles were used for the growth of each passivation layer. The ZnO shell thickness was around 2 nm, which was characterized by the growth of ZnO on dummy Si substrates. The positions of the nanowires were mapped with a microscope, and then, electrical contacts on both ends of the nanowires were patterned by HEIDELBERG DWL66+ maskless photolithography. After that, ZnO at the contact area was removed, followed by deposition of Cr/Au (20/80 nm) electrodes. After a lift-off process in acetone, the electrical and optoelectric responses of GaAs nanowire devices were characterized using a semiconductor parameter analyzer. A light power meter was used to measure the illumination intensity.

### 3. Results and discussion

During the initial stage of ZnO growth by ALD, the deposition rate per cycle could be less than a monolayer. The growth mode could be attributed to the induction time for the nucleation of ZnO nuclei on a foreign substrate.<sup>34</sup> Furthermore, *in situ* X-ray fluorescence data for ZnO ALD grown on an InGaAs substrate verified that Zn desorption from the substrate surface should be responsible for the extremely slow growth rate (<0.05 nm per cycle) during the first several cycles.<sup>35</sup> Consequently, the ALD-ZnO process on the GaAs nanowires for only a few cycles is considered to be a surface treatment instead of a physically complete and distinct layer.

The  $\mu$ -PL intensity of intrinsic GaAs NWs could indicate the relative density of the surface states, because the increase in PL intensity is related to the lower recombination velocity. The intrinsic GaAs nanowires could behave as weakly p-doped semiconductors, because of native acceptor defects, such as Ga vacancies. To avoid PL signal interference from the silicon substrate, individual nanowires were transferred onto the quartz flake. PL spectra were measured under an excitation laser with 532 nm wavelength at room temperature. As shown in Fig. 1, an average 20-fold increase in the peak intensity for the individual nanowires passivated by ALD was obtained, indicating a significant reduction in surface recombination velocity. Considerable passivation results are within expectations, because nearly complete Fermi level depinning from the valence band edge can be achieved for p-GaAs passivated by ALD ZnO.<sup>31</sup> A higher doping level could cause a thinner surface depletion area; hence the effect of the surface states is more severe for the intrinsic GaAs nanowires than that for doped ones.

The PL quenching could be a result of the type-II band alignment between the passivation shell and inner semiconductor materials, which is related to the efficient separation of photogenerated carriers and effective reduction in radiative recombination. However, this normally happens with

thick shells. In our case, the passivation layer is ultra-thin. To the best of our knowledge, there is still the lack of a report about a band alignment diagram between ultrathin ZnO and GaAs, but it is believed to be difficult to form a clear type-II band alignment. Here, ALD ZnO efficiently suppresses oxide formation and minimizes Fermi-level pinning. The greatly improved PL intensity demonstrates that the passivation effect is dominant and PL quenching by type-II band alignment is not the main concern in this structure.

As shown in Fig. 2(a), Cr/Au electrodes are deposited on the two ends of a single p-type GaAs nanowire, and the heavily p-doped silicon substrate serves as the back gate. The SEM image shows a single NW-based device (Fig. 2(b)). The current becomes larger with source-drain bias with bending characteristics, caused by a back-to-back Schottky configuration. Although the back gate modulation role is insufficient, as displayed in Fig. 2(c), the output characteristics under the illumination of a microscope still confirm the p-type conductivity of a GaAs nanowire. The gate voltage varies from  $-30$  V to  $+30$  V with a step of 10 volts. The accumulation of holes inside the nanowire channel occurs under a negative gate voltage, which raises the drain current. The case of positive gate bias tends to deplete majority carriers, which reduces the drain current.

The existence of ZnO on the substrate can be confirmed by XPS Zn 2p spectra whose peak is located at about 1022 eV, and this peak relates to the Zn–O bond (Fig. 2(d)). Characterization of the presence of ZnO was also performed by PL and Raman tests, but they do not work for capturing such a weak signal, due to its ultrathin nature. The chemical etching behavior of ZnO could refer to reference,<sup>36</sup> which shows that the etching rate of ZnO by HCl is very fast. In this article, ZnO present in the source/drain part is etched away by diluted hydrochloric acid.

The dark current  $I_d$  without gate bias under forward- and reverse-biased voltages is slightly different, which may originate from unequal Schottky barrier heights at the two ends. Compared with the reference device in our previous report,<sup>33</sup> its dark current increases at the same bias voltage. A similar phenomenon, referred to as conductivity improvement, has been found in previous literature,<sup>18,37–39</sup> determined by reduction in surface trap density.

The surface states on p-GaAs nanowires act as recombination centers for free carriers and positively charged surface traps.<sup>40</sup> A depletion shell exists and the effective channel available for carrier transport is narrowed and confined, because of the surface states.<sup>41</sup> XPS spectral analysis, which could reflect interfacial chemical bonding, was utilized to evaluate the results on (In)GaAs substrates by an ALD ZnO passivation layer.<sup>31,32</sup> A substantial decrease in As–As bonding and a drop in Ga–O bonding were observed. The strong Zn–O bond strength induces a nearly impossible reaction between oxygen atoms in ZnO and Ga(As) atoms,<sup>42</sup> and the conformal dense coverage of ZnO effectively suppresses the penetration of oxygen from the atmosphere into NWs. Besides, the Au/Zn/Au structure is one option for GaAs Ohmic contact, as Zn could serve as a p-type dopant.<sup>43</sup> Hence, suppression of nanowire

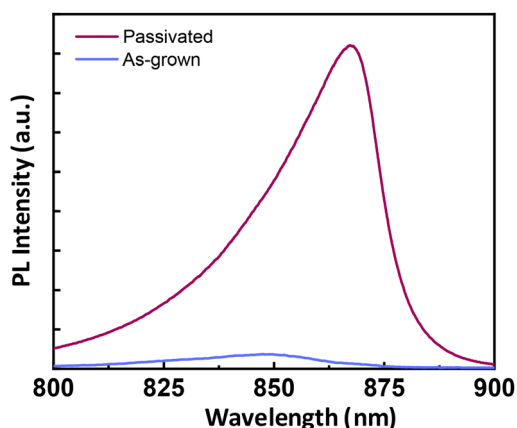
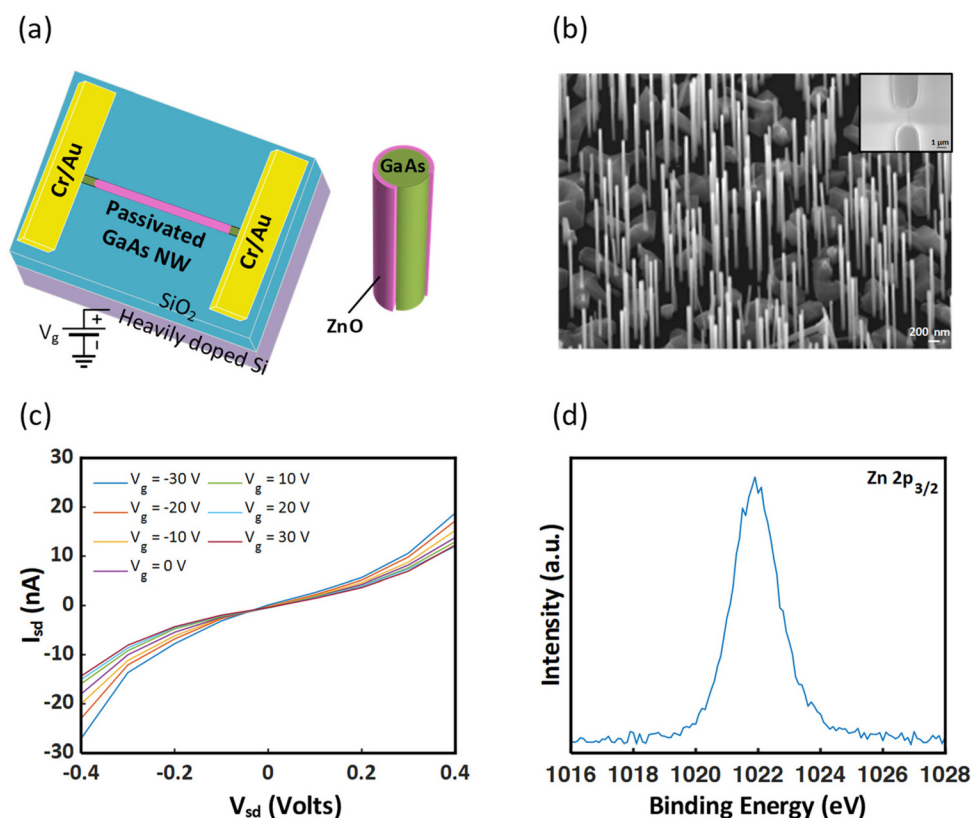


Fig. 1 Comparison of average micro-PL spectra between passivated intrinsic nanowires and pristine intrinsic nanowires.



**Fig. 2** (a) Device schematic. (b) SEM image of the GaAs nanowires. (The inset shows the SEM image of the GaAs nanowire photodetector.) (c) Output characteristics of NW FET under different gate biases. (d) XPS Zn 2p spectra.

conductivity induced by Fermi level pinning could be alleviated in part, *via* the introduction and diffusion of Zn atoms. The improvement in the conductivity of photodetectors enables better photocurrent transportation, making a boost in photoresponsivity possible.<sup>44</sup>

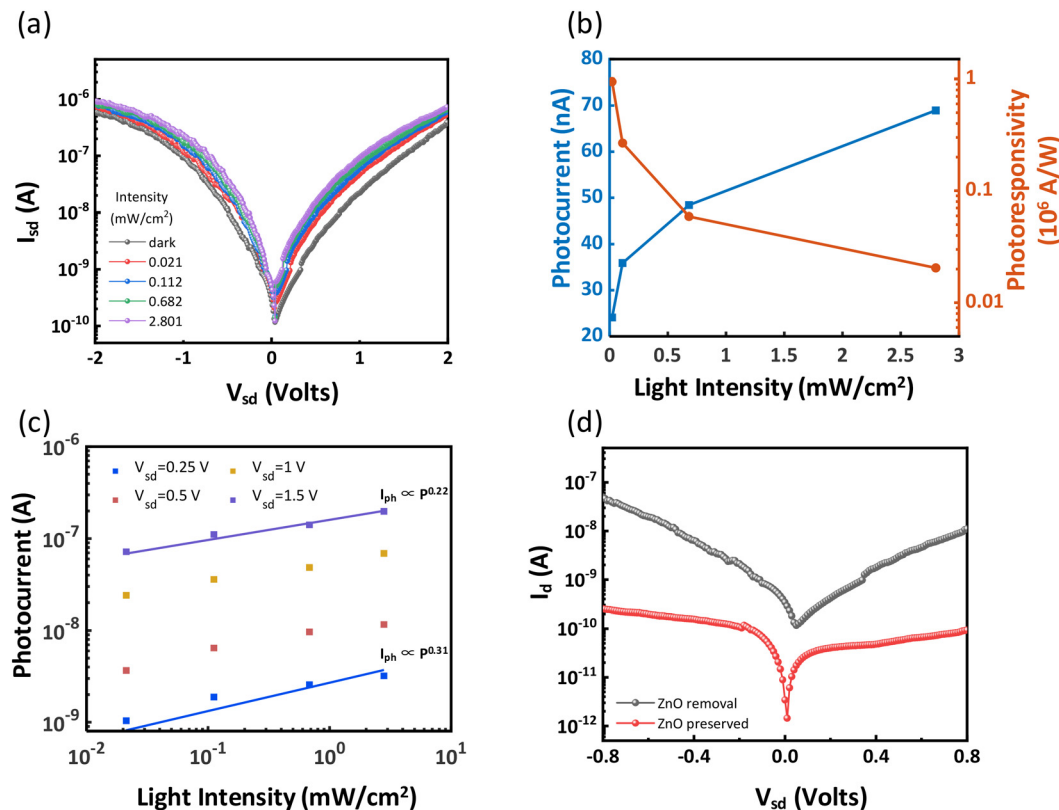
The optoelectronic performance of the modified device without gate voltage is plotted in Fig. 3. The net photocurrent is defined by  $I_{ph} = I_{light} - I_d$ , where  $I_{light}$  denotes the current under illumination. The device shows a distinct photocurrent even under a very weak illumination. Generally, the current increases as illumination intensity becomes larger. The photoresponsivity is defined by  $R = I_{ph}/PA$ , where  $P$  means the incident light intensity per unit area, and  $A$  is the effective illuminated area associated with nanowire geometric size and channel length. The spectral responsivity rises with increasing bias. Additionally, the specific detectivity is calculated with the formula  $D^* = R(A/2eI_{dark})^{1/2}$ , where  $e$  is the electron charge. Moreover, another vital parameter named the optical gain, which is related to the electron-hole pair quantity collected by the electrodes upon excitation by one incident photon, is defined as  $G = R \times hc/\lambda e$ , where  $h$  denotes the Planck constant,  $c$  is the speed of light, and  $\lambda$  means the wavelength of the irradiation light. The relationship of responsivity *versus* incident power is plotted in Fig. 3(b). It is clearly shown that the photoresponsivity drops with increasing incident power

density, and the variation trend of the specific detectivity is the same. These two parameters can reach  $946 \text{ kA W}^{-1}$  and  $3.93 \times 10^{14} \text{ Jones}$ , respectively, when illuminated by  $21 \text{ } \mu\text{W cm}^{-2}$  532 nm light.

To obtain a further understanding of the photodetector performance, the working principle of the device is discussed here. Limited by the Fermi level pinning at one-third of the bandgap above the valence band maximum, the Schottky barrier height is usually fixed regardless of the metal work function. The difference in negative work function between Cr and p-type GaAs leads to downward band-bending at the GaAs NW/metal contact interface. A layer of negative charges forms in the GaAs side, while a layer of positive charge is built in the metal region. One Schottky junction is forward-biased, just acting as a resistor in series, while the other is reverse-biased on which most of the voltage drop is applied. A built-in electric field inside the reverse-biased Schottky junction could suppress the recombination rate of photon-generated electron-hole pairs or separate the photon-generated electron-hole pairs efficiently.<sup>39</sup> The Schottky barrier height could be extracted from the following formula:

$$e\Phi_B = \frac{kT}{e} \ln \left( \frac{A^* T^2}{J_s} \right),$$





**Fig. 3** Optoelectronic performance at zero gate bias (a) output characteristics of a nanowire photodetector under 532 nm wavelength light for different intensities. (b) Dependence of the photocurrent and responsivity on incident illumination density at 1 V bias voltage. (c) Variation of photocurrent with light intensity for various bias voltages. (d) Dark current comparison between the devices with and without ZnO in the source/drain part.

where  $k$  is the Boltzmann constant,  $T$  denotes the temperature in kelvin,  $J_s$  means the saturation current density, and  $A^*$  represents the effective Richardson constant (for p-type GaAs it is  $74 \text{ A K}^{-2} \text{ cm}^{-2}$ ). The Schottky barrier height is estimated to be about 0.47 eV, which shows a drop relative to the pristine device. Benefiting from a lower Schottky barrier height upon channel passivation, more efficient carrier collection is allowed. Consequently, amplification of dark current and photocurrent could occur on the same device. It is worth noting that the efficient reduction of the surface states enables better charge carrier transportation inside the NW channel and therefore decreases the loss of photo-generated carriers, which could ensure a good collection of photo-generated carriers at the electrodes. A more conductive channel is responsible for a more pronounced photocurrent and photo-responsivity with a large dark current. Apart from this, optically excited free excess holes are able to lift the Fermi energy of the hole gas within the nanowires, which is called the photodoping effect that could raise the conductance of semiconductor circuits.<sup>48</sup>

The peak of the photo-response ratio expressed by  $I_{ph}/I_d$  is about 12.56, under 780 nm light with a power of  $3 \text{ mW cm}^{-2}$ . It becomes large with an increase in light power, due to the

creation of more photo-generated carriers. The light-intensity-dependent photocurrent under different biases is displayed in Fig. 3(c) and could be fitted using a power law,  $I_{ph} \propto P^\alpha$ , where the extracted component  $\alpha$  could indicate the type of photogeneration mechanism. The photocurrent increases with applied bias. Generally speaking, the photocurrent is dominated more by the photoconductive effect when the component approaches 1, and a sub-linear relationship indicates the presence of some complex photogeneration and recombination processes.<sup>49</sup> The less depleted channel produces a smaller  $\alpha$  value, which is consistent with other reports.<sup>47</sup> The decrease in  $\alpha$  dependence on bias is possibly induced by the effect of defects on carrier transportation.<sup>50</sup> Stronger illumination induces larger carrier concentration near the Schottky area, resulting in a reduced built-in electric field and weakened carrier separation efficiency. Limited to the small effective illuminated area of a single nanowire and the inherent property of a single GaAs nanowire device, it is hard to get a very distinct photo-response ratio like other materials, which is consistent with other papers on GaAs nanowire photodetectors.<sup>33,44,47</sup>

$I$ - $V$  measurements were conducted on a nanowire device without removal of ZnO before metal contact deposition, as

**Table 1** Performance comparison of GaAs(Sb) NW-based photodetectors with previous work

Material	Wavelength (nm)	Bias (V)	Dark current	Photoresponsivity ( $\text{A W}^{-1}$ )	Specific detectivity (Jones)	Optical gain (%)	Ref.
Si-doped GaAs NW	532	−1	~50 pA	1175	$3.02 \times 10^{12}$	$2.74 \times 10^5$	44
Thiourea-passivated GaAsSb NW	1550	2	1.3 $\mu\text{A}$	$5.4 \times 10^4$	$2.5 \times 10^{10}$	$4.4 \times 10^6$	45
Sulfur-passivated undoped GaAs NW	532	1	0.07 pA	25	$9.04 \times 10^{12}$	—	46
Sulfur-passivated p-GaAs NW	520	2	22.2 $\mu\text{A}$	$6.4 \times 10^5$	$8.2 \times 10^{12}$	$1.5 \times 10^8$	47
p-GaAs NW (reference device)	632.8	−5	4 nA	$1.45 \times 10^5$	$1.48 \times 10^{14}$	$2.85 \times 10^7$	33
<b>Passivated p-GaAs NW</b>	<b>532</b>	<b>1</b>	<b>21.97 nA</b>	<b><math>9.46 \times 10^5</math></b>	<b><math>3.93 \times 10^{14}</math></b>	<b><math>2.2 \times 10^8</math></b>	<b>This work</b>

shown in Fig. 3(d), and one possible source is responsible for a low current. The existence of an energy barrier caused by an interface valence band offset hinders carrier collection by metal electrodes.

A comparison of device performance is given in Table 1. Since the passivation alleviates the surface states and leaves more free photocarriers, the role of the Schottky junctions in the photogeneration mechanism may thus be reasonably dominant in photoresponse enhancement. The passivated device exhibits ultrahigh and broadband responsivity to weak light, considerable detectivity, and a normal photoresponse ratio. This superior behavior could be attributed to three factors: (1) the high-quality crystal of self-catalyzed GaAs nanowires ensures lower numbers of nonradiative recombination centers and carrier traps. (2) The ZnO passivation could effectively reduce the surface states present in GaAs and improve conductance, enabling better photocurrent transportation. (3) A reverse-biased Schottky junction broadens the barrier width and strengthens the built-in electric field, which promotes the separation of photo-generated carriers and results in high photoresponsivity.

## 4. Conclusions

In summary, the passivation efficacy of ZnO deposition on the intrinsic GaAs nanowires was investigated, and a surface-modified high-quality doped GaAs nanowire photodetector with outstanding performance was demonstrated. The PL enhancement and increase in conductance after ZnO treatment prove its vital role in reducing surface recombination velocity and surface state density as a surface passivation treatment. The dramatic elimination of the surface states upon passivation ensures good photoconductance, and the low growth temperature of ZnO allows such a passivation procedure, which is more compatible with modern semiconductor device processing techniques. Compared to the pristine device in our previous report, the photoresponsivity is found to be extraordinarily enhanced up to  $946 \text{ kA W}^{-1}$ , and it shows a considerable specific detectivity under 532 nm light. This device configuration opens a new gateway for enhancing photodetector performance with high responsivity, detectivity and optical gain, which is expected to be extended to other types of low-dimensional semiconductor materials.

## Author contributions

All authors contributed to the writing of the manuscript. Additionally, Fuxiang Shang: conceptualization, investigation, formal analysis, writing original draft; Chao Fei Zha: investigation, formal analysis; Hanchen Zhu: investigation, data curation; Zheyu Zhang: methodology; Yuanhao Shen: investigation; Qichao Hou: investigation; Linjun Zhang: conceptualization; Yanmeng Chu: data curation; Lulu Chen: investigation; Junjie Zhao: resources; Wenzhang Fang: resources; Huiyun Liu: resources; Yishu Zhang: resources; Zhiyuan Cheng: supervision, funding acquisition, project administration; Yunyan Zhang: supervision, funding acquisition, project administration.

## Conflicts of interest

There are no conflicts to declare.

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