



Cite this: *Polym. Chem.*, 2024, **15**, 4319

Scalable design of uniform oligourethanes for impact study of chain length, sequence and end groups on thermal properties†

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The full potential of sequence-defined macromolecules remains unexplored, hindered by the difficulty of synthesizing sufficient amounts for the investigation of the properties of such uniform structures and their derived materials. Herein, we report the bidirectional synthesis and thermal behavior analysis of sequence-defined oligourethanes. The synthesis was conducted on a large scale (up to 50 grams) using a straightforward protocol, yielding uniform macromolecules as validated by NMR, ESI-MS and SEC. With this approach, a library of uniform oligourethanes (up to the octamers) was produced using two structural units: a hydrogen-bonding carbamate and a methyl-substituted alternative structure. By varying the chain length, monomer sequence and functionality, we were able to perform a systematic study of the impact of hydrogen bonding on the thermal properties of polyurethanes. Thermal analysis of the discrete oligomers using DSC revealed that both the molecular weight and microstructure significantly affect the glass transition and melting temperatures. TGA measurements also revealed differences in the thermal stability of the oligomers, underscoring the significance of the primary structure of polyurethanes. Additionally, the influence of the terminal groups on the degradation pathway was assessed via pyrolysis-GC-MS, which specifically highlighted the increased thermal stability in the absence of hydroxyl end groups. This work shows the interest of using sequence-defined synthetic macromolecules for the elucidation of structure–property relationships and thereby facilitates their fine-tuning towards specific material applications.

Received 10th September 2024,
Accepted 9th October 2024

DOI: 10.1039/d4py01001a

rsc.li/polymers

Introduction

Hydrogen bonds are an important class of non-covalent interactions that can greatly impact the performance of a polymeric material. A prime example of this is spider silk, one of nature's most resilient protein fibers, which possesses an exceptional strength while maintaining high deformation.^{1,2} This extraordinary behavior arises from the hierarchical structure of the material, which comprises glycine-containing amorphous regions and alanine-rich domains. The aggregation of the latter into crystalline regions adds mechanical strength and hardness through physical crosslinks, a phenomenon that is responsible for the high strength of polyurethanes (PUs), one of the most widely used commodity plastics.³ The application of PU-based materials ranges from soft foams and mattresses

to thermal insulation for buildings, leading to an estimated annual consumption of about 20 million metric tons.⁴ The popularity of PUs is driven by their superior and variable mechanical properties compared to alternative polymers, a characteristic that is primarily linked to their strong structure–property relationships.^{3,5–7} Indeed, within their structure, the segregation of the hard segments, which are crystalline regions formed by the strong hydrogen bonds between carbamate and urea groups, plays a crucial role in influencing the material's behavior. Given the vital role of these non-covalent interactions in dictating material properties, numerous efforts have been dedicated to the study of their thermal and material properties.^{8–13} However, due to the inherent dispersity resulting from both the applied polyols and the step-growth polymerization process, extracting exact fundamental information on the structure–property relationship proves challenging.

Recently, the issue of dispersity has been addressed through the introduction of sequence-defined macromolecules in polymer science.^{14–16} Aiming to mimic nature's biopolymers, monodisperse macromolecules with a defined monomer order were developed, targeting applications where the primary structure is of utmost importance, such as catalysis, self-assembly and data-storage.^{17–26} Such uniform polymers

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† Electronic supplementary information (ESI) available: Details about instrumentation, synthesis and experimental procedures, NMR-, SEC-, ESI-MS- and orbitrap-MS spectra, as well as DSC and TGA thermograms. See DOI: <https://doi.org/10.1039/d4py01001a>



allow for a thorough examination of structure–property relationships, which cannot be properly achieved by using conventional polymers or polymers derived from reversible-deactivation radical polymerizations (RDRPs), due to their dispersity. Several research groups have already demonstrated this potential in their studies.^{19,27–30} For example, the group of Alabi investigated the duplex formation of sequence-defined oligomers and found that the assembly is highly dependent on the positioning and valency of the complementary hydrogen binding partners.²⁷ Our own research group has also provided an example where the structure-dependent binding affinity has been clearly demonstrated for uniform antibody-recruiting macromolecules.²⁸

The synthesis of these discrete polymers involves iterative protocols, progressively expanding the polymer chain by one unit after each cycle. Throughout these iterations, it is essential to ensure that all steps reach complete conversion to prevent the emergence of dispersity or irregularities in the sequence, which is often achieved by choosing efficient chemistries and using an excess of reagents. Since the remaining reagents may disrupt subsequent couplings, it is imperative to conduct a work-up after each step. The introduction of solid supports significantly simplifies this purification, requiring only a straightforward filtration and washing step to remove all waste products effectively.^{31,32} However, some drawbacks of this approach include the high cost of the resins, their restricted loading capacity, and the need for a large excess of often expensive reagents, which hamper the scalability of the synthesis. Transitioning from a solid to a soluble support addresses several of these challenges by reducing the excess reagents and costs related to the support, while simultaneously providing more homogeneous reaction conditions. Furthermore, this approach also allows for direct characterization without the need for chain cleavage, and the purification is limited to precipitation.^{33,34} Nevertheless, a supportless syn-

thesis remains the most advantageous approach due to its increased atom efficiency and the elimination of the often cumbersome support synthesis. Recently, this strategy was exploited by our group and others for the synthesis of sequence-defined oligourethanes.^{17,35–37}

In this work, the approach has been adapted to synthesize these uniform macromolecules in a scalable manner from aminoethanol and using a coupling agent. Not only has the protocol been modified to allow a bidirectional growth of the oligomer chain but the purification is also limited to a single extraction or precipitation step (Fig. 1). Two structural units have been selected in this study, *i.e.* one containing a regular N–H carbamate unit with hydrogen bonding capacity, and one methyl-substituted counterpart that lacks this capability. A detailed exploration of the structure–property relationships of a series of oligomers with varying chain length, sequences, and end-group functionalities was performed to provide an in-depth understanding for the precise impact of hydrogen bonding on polyurethane thermal characteristics. By exploring the interplay between molecular design and thermal properties, this study paves the way for tailoring material performance to specific application requirements.

Results and discussion

Synthesis of oligourethanes with distinct primary structure

Several oligourethanes with distinct primary structures were synthesized using the protocol outlined in Fig. 1. By selecting 1,10-decanediol as the initiator, bidirectional growth of the oligomer chain was achieved, which accelerates the oligomerization process and opens up possibilities for telechelic oligomer synthesis. Furthermore, this procedure facilitates the purification of the oligomers by increasing their hydrophobicity thanks to the long aliphatic chain. The iterative protocol,

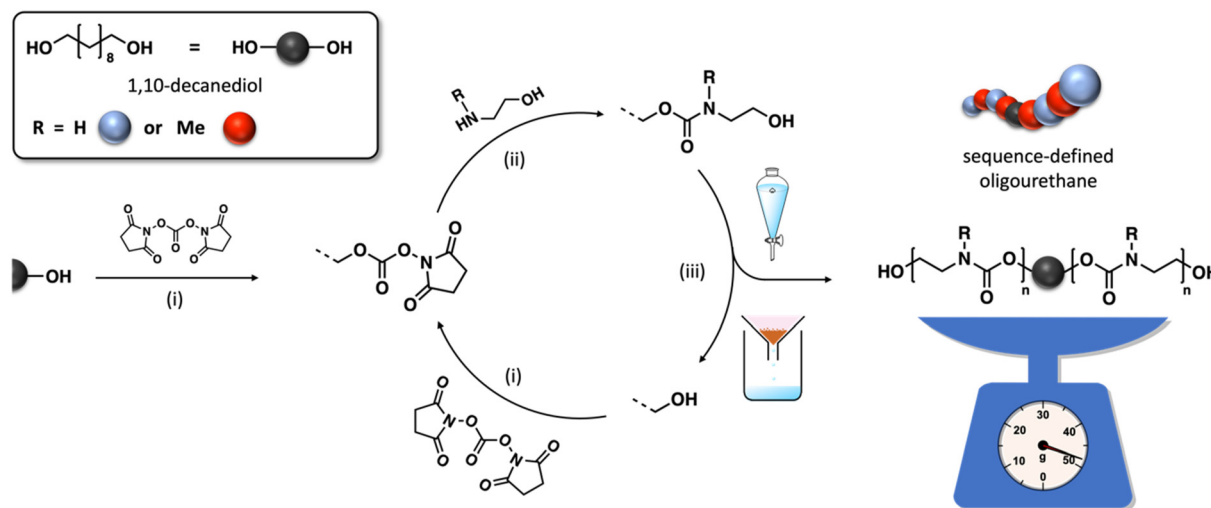


Fig. 1 General synthesis protocol for the synthesis of uniform oligourethanes consisting of (i) an activation of the terminal alcohol groups to a carbonate, (ii) the subsequent chemoselective coupling with an aminoethanol and (iii) the purification with either an extraction or a precipitation.



carried out in acetonitrile (ACN) as the solvent, starts by converting the terminal hydroxyl groups of the initiator into *N*-hydroxysuccinimide (NHS)-containing carbonates by making use of disuccinimidyl carbonate (DSuC) in slight excess (*i.e.*, 1.25 equiv. per OH group). Subsequently, an alkanolamine is introduced that chemoselectively forms carbamate bonds, while simultaneously regenerating terminal alcohol moieties. In this step, the choice of primary or secondary amine determines whether the incorporated monomer can form hydrogen bonding interactions or not. Given that some DSuC remains from the first step, an intermediate work-up would be required to prevent side reactions. However, this additional step can be avoided by introducing the alkanolamine in excess (*i.e.*, 4 equivalents) in the second step. This excess leads to the quenching of the remaining DSuC without causing side reactions that could interfere with the growing sequence. Using this initiator and 2-(methylamino)ethan-1-ol, four OH-endcapped methylated oligourethanes (**M_x-OH**) with different molecular weights (see Table 1) were synthesized: a dimer (**M₂-OH**), a tetramer (**M₄-OH**), a hexamer (**M₆-OH**) and an octamer (**M₈-OH**). All these products were obtained on a multigram scale (23 g of **M₂-OH**, 33 g of **M₄-OH**, 42 g of **M₆-OH**, and 50 g of **M₈-OH**) with high yields (95% per step), and their purity was confirmed by ESI-MS, SEC and NMR (Fig. S1–S26†). To demonstrate that the protocol can be performed on larger scale, the synthesis of **M₈-OH** was successfully repeated on a 50 gram scale.

The oligomers were purified by means of an aqueous extraction, successfully removing all water-soluble side products. This process is made possible by the non-solubility of the methyl-substituted oligourethanes in water and their solubility in non-miscible organic solvents, such as dichloromethane (DCM), improved by the long aliphatic chain of decanediol. Earlier attempts with ethanediol resulted in a more challenging extraction and low yields after purification (<50%, data not shown).

In parallel, four monodisperse oligomers (**H_x-OH**), having similar structures but without methyl-substituents, have been

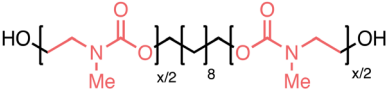
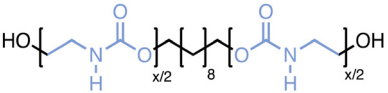
prepared using 2-aminoethan-1-ol this time. While the **M_x-OH** oligomers containing substituted carbamates could be isolated using extraction, the **H_x-OH** oligomers were insoluble in any organic solvent. Remarkably, polar solvents like methanol, ethanol, isopropanol, and water did not allow solubilization of those structures, whereas only dimethyl sulfoxide (DMSO) and dimethyl formamide effectively disrupted the non-covalent interactions, thereby dissolving the product. As a consequence, the solvent used in the protocol was switched from ACN to DMSO and the work-up approach adjusted accordingly. For the latter, the entire reaction mixture was added to water and stirred vigorously to dissolve all byproducts, with the oligourethane remaining as a white powder that could be filtered off. Because filtration is a straightforward and scalable work-up method, this alternative procedure does not compromise any efficiency compared to the conventional extraction process. This was demonstrated by synthesizing a similar series of oligourethanes, including a dimer (**H₂-OH**), tetramer (**H₄-OH**), hexamer (**H₆-OH**) and octamer (**H₈-OH**), in high yields per step (*i.e.*, 97%). Moreover, the monodisperse octamer synthesis was performed on a 10 gram scale and the purity of these oligomers has been confirmed again by ESI-MS, SEC and NMR (see Fig. S1–S26†).

Since these sequence-defined oligourethanes vary in chain length, further investigations could be conducted on the effect of this parameter on their thermal properties including glass transition temperature (*T_g*), thermal degradation and melting behavior.

Influence of the length

Starting with the methylated carbamate series (**M_x-OH**), SEC analysis shows narrow peaks, as expected from the uniformity of the macromolecules, along with a clear shift that demonstrates the consistent increase in molecular weight with each monomer addition (Fig. 2). When examining their thermal behavior using differential scanning calorimetry (DSC), only a single thermal transition is observed, confirming their amor-

Table 1 Targeted oligourethanes consisting of (top) Me-substituted- (**M_x-OH**) and (bottom) regular- (**H_x-OH**) carbamates, along with their corresponding monoisotopic masses and thermal properties

Sequence-defined oligourethane	<i>x</i> =	<i>m/z</i> _{th} ^a	<i>m/z</i> _{exp} ^b	<i>T_g</i> ^c (°C)	<i>T_m</i> ^d (°C)	Δ <i>H</i> _{fus} ^e (kJ mol ^{−1})
 M_x-OH	2	377.2646	377.2641	−66	—	—
	4	579.3600	579.3602	−44	—	—
	6	781.4553	781.4552	−32	—	—
	8	983.5507	983.5529	−24	—	—
 H_x-OH	2	349.2333	349.2331	—	120	70
	4	523.2974	523.2977	—	136	107
	6	697.3614	697.3612	—	147	130
	8	871.4255	871.4235	—	150	151

^aTheoretical monoisotopic *m/z*. ^bDetermined monoisotopic *m/z* via orbitrap-MS measurements. ^cGlass transition temperature determined via the second heating run in DSC. ^dMelting temperature determined via the first heating run in DSC. ^eEnthalpy of fusion integrated from the first heating run in DSC.



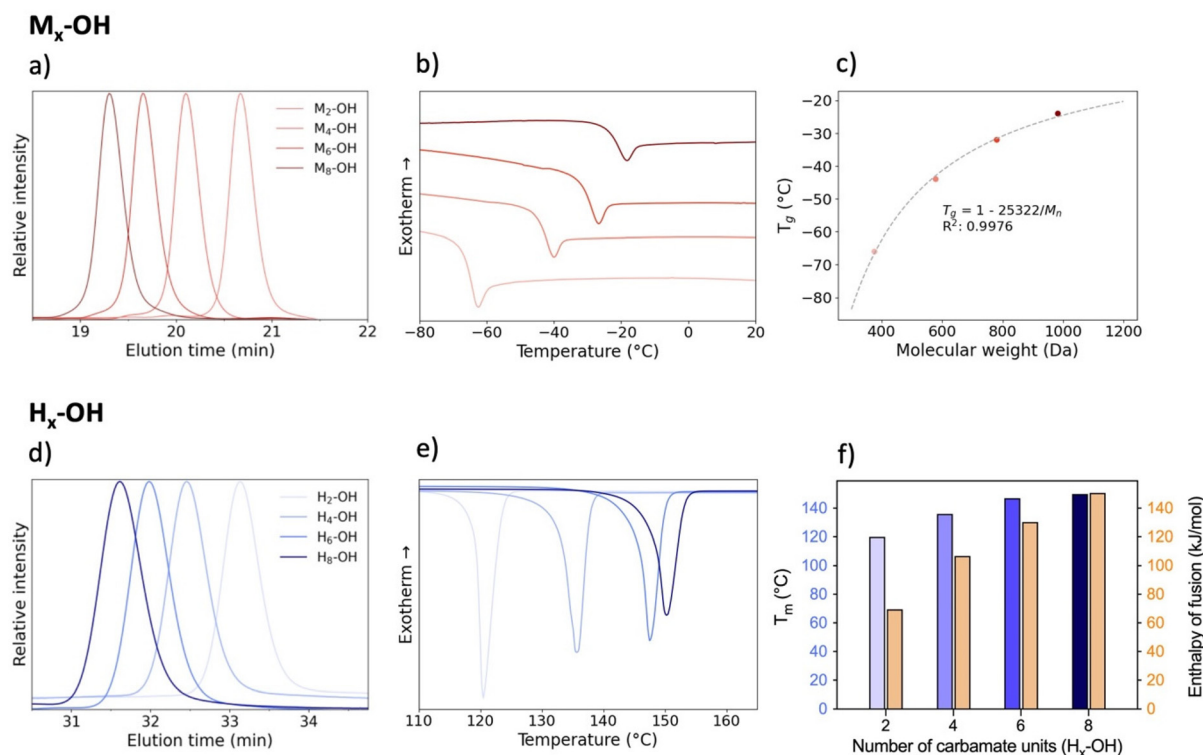


Fig. 2 Characterization of the M_x -OH (top) and H_x -OH (bottom) oligomers. (a and d) SEC traces of the various macromolecules. (b and e) DSC curves showing the evolution of T_g or the melting temperature when varying the chain length for oligomers M_x -OH or H_x -OH, respectively. (c) Flory-Fox fitting showing the relationship between the T_g and the molecular weight. (f) Bar plots of the melting temperatures and enthalpy of fusion.

phous state. This T_g increase can be correlated to the polymer's molecular weight *via* the Flory-Fox equation (eqn (1)).³⁸ This equation includes two polymer specific parameters: $T_{g,\infty}$, which denotes the highest possible T_g achievable at a theoretical infinite molecular weight, and K , an empirical parameter related to the free volume present in the polymer sample. Fitting the values to this equation yields a $T_{g,\infty}$ of 1 °C and a K of 25 322 kg mol⁻¹, supported by a high R^2 value, demonstrating the suitability of sequence-defined polymers for elucidating structure-property relationships.

$$T_g = T_{g,\infty} - \frac{K}{M_n} \quad (1)$$

The oligomers with hydrogen bonding moieties (H_x -OH) exhibit no T_g in the DSC-analysis due to their high degree of crystallinity. On the other hand, the melting peak gives information about the melting point and energy required to put the white powders into their liquid phase. As expected, extending the chain length and, consequently, increasing the number of hydrogen bonds results in a higher melting point. Furthermore, the enthalpy of fusion (ΔH_{fus}) increases with the chain length, with a value of about 70 kJ mol⁻¹ for H_2 -OH up to about 150 kJ mol⁻¹ for the octamer H_8 -OH. This clearly shows that for such low mass oligourethanes, the amount of N-H containing carbamates has a significant impact on the

crystallinity of the polymers and consequently on their thermal behavior.

Influence of sequence and end group functionality on the thermal behavior of sequence-defined oligourethanes

These first experimental results raised the question whether the transitions of such oligourethanes could be precisely controlled when combining regular carbamates with substituted carbamates. Specifically, we aimed to assess the influence of the sequence on the resulting material properties. To study the impact of monomer arrangement, the library of sequence-defined oligourethanes was further expanded (see Fig. 3) to include both a blocky structure (**B₈-OH**), featuring a substituted urethane domain in the center, flanked by N-H urethane segments, and an alternating structure (**A₈-OH**). Since the synthesis of these copolymers did not result in any solubility issues thanks to the insertion of a methyl-substituted monomer as the first unit, the protocol was carried out in a manner similar to the **M₈-OH** octamer (see Experimental section). Even though the molecular weights and number of non-substituted carbamates are identical, the varying synergistic effects exerted by these hydrogen bonding groups can lead to differences in thermal behavior. Furthermore, the influence of the terminal groups is also examined by replacing the OH end groups of the oligomers with alkene moieties. The latter, chosen for the potential introduction of such oligomers



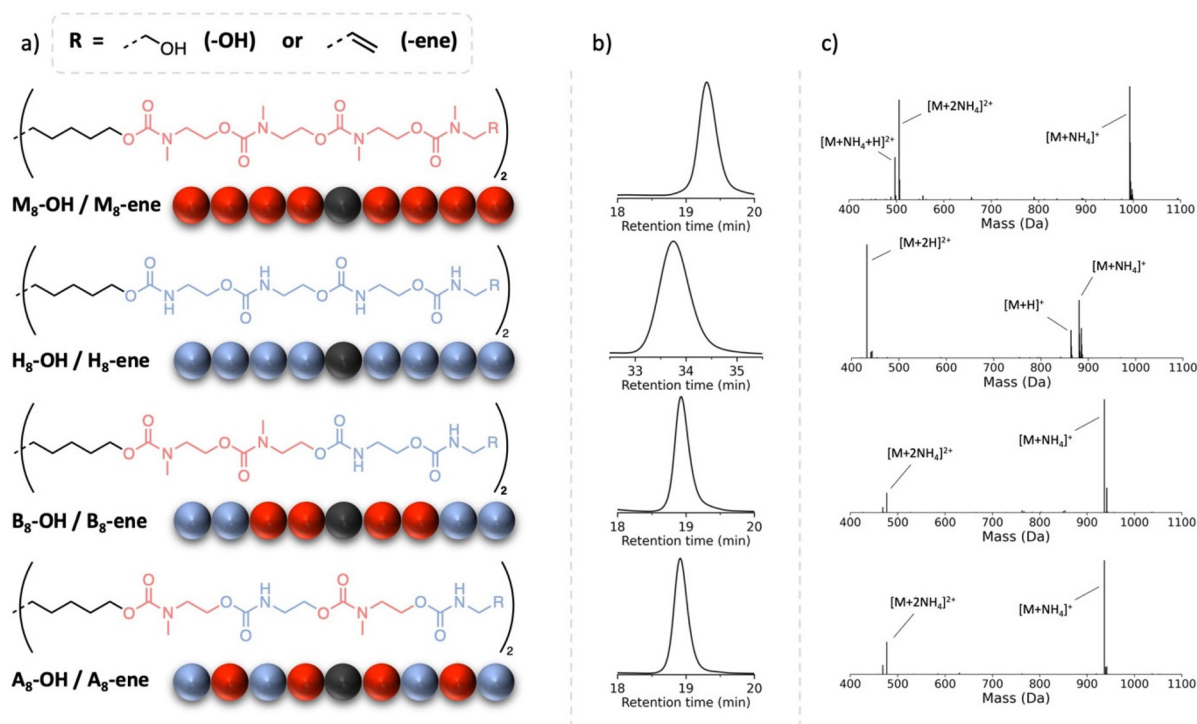


Fig. 3 (a) The different structures synthesized, from top to bottom: methylated ($M_8\text{-OH}/M_8\text{-ene}$), non-methyl substituted ($H_8\text{-OH}/H_8\text{-ene}$), block ($B_8\text{-OH}/B_8\text{-ene}$) and alternating ($A_8\text{-OH}/A_8\text{-ene}$) oligomers with hydroxyls (–OH) or alkene (–ene) end groups. (b) SEC traces and (c) ESI-MS of the corresponding alkene-terminated oligomers. Note that the second SEC measurement has been performed in dimethylacetamide due to a solubility issue. This SEC system utilizes a column calibrated for higher molecular weight polymers, resulting in the oligomers having a later retention time compared to the other measurements.

into crosslinked structures, is achieved by exchanging the last added alkanolamine with the corresponding allylamine. This procedure gives rise to four new oligomers, **H₈-ene**, **M₈-ene**, **B₈-ene** and **A₈-ene**.

All eight oligomers were analyzed using DSC (see Fig. 4a and Table 2), with T_g values obtained from the second heating run. On the other hand, T_m along with melting enthalpy values – if present – were recorded from the first heating run because of a very slow recrystallization process that could not be fully achieved, even with very low cooling rates. With the exception of the highly crystalline, non-methyl-substituted oligourethanes (**H₈-OH** and **H₈-ene**), all oligomers demonstrate a clear T_g transition. Efforts have been made to enable the identification of the T_g of **H₈-OH** and **H₈-ene**, such as annealing these products and performing dynamic mechanical analysis (DMA), yet no results have been obtained. Focusing first on the influence of the end group, varying effects on the T_g can be observed. In purely methyl-substituted oligomers, terminal hydroxyl groups slightly raise the T_g from -34°C (**M₈-ene**) to -24°C (**M₈-OH**), whereas in the block and alternating copolymers, these OH groups lead to a decrease of the T_g (e.g., $T_g = -42^\circ\text{C}$ for **B₈-OH** vs. $T_g = -13^\circ\text{C}$ for **B₈-ene**). A potential explanation is that the OH groups in **M₈-OH** introduce additional intermolecular interactions, increasing the T_g , while the hydroxyl groups in the copolymers (**B₈-OH** and **A₈-OH**) lower the T_g by interfering with the existing hydrogen bonds. The

sequence also has a pronounced effect on the T_g , as is clearly demonstrated by the differences between the ene-terminated oligomers. Indeed, it was found that **B₈-OH** and **A₈-OH** possess similar T_g values (-11°C and -13°C , respectively), whereas **M₈-OH** exhibits a significant decrease in its T_g value (-34°C). This highlights the T_g -lowering effect of carbamate substitution (see Fig. 4a), as those substituted moieties do not reduce the molecular chain mobility in a significant way.

While **H₈-OH** and **H₈-ene** cannot be compared to the other oligomers in terms of their T_g values, they can be studied in terms of T_m and ΔH_{fus} . It was anticipated that substituting the OH group with an alkene would lower the melting temperature due to reduced non-covalent interactions. However, contrary to this expectation, the diene-terminated octamer demonstrates higher values for both the melting temperature and enthalpy of fusion. While all methyl-containing oligomers are amorphous and thus do not exhibit this transition, **A₈-ene** is an exception. It slowly crystallized from a viscous oil to a white powder with a melting temperature of 70°C . This confirms the strong structure–property relationship in oligourethanes, which can be accurately investigated with sequence-defined macromolecules, in contrast to disperse oligomers and polymers that only allow for averaged property determination.

Additionally, thermogravimetric analysis (TGA) was performed to gain insight into the impact of the microstructure on the thermostability of these sequence-defined oligomers. In



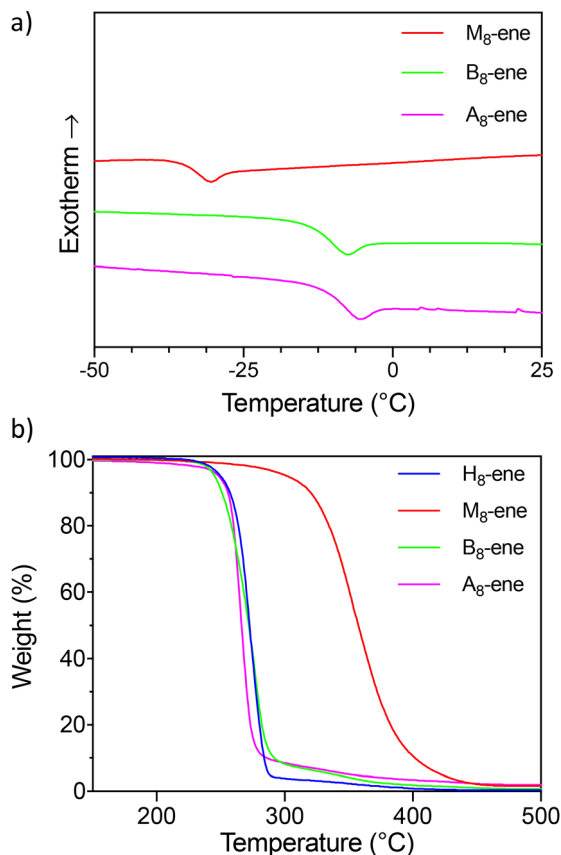


Fig. 4 (a) DSC measurements of the alkene-terminated octamers showing their T_g . (b) TGA measurements of the alkene terminated octamers.

this context, differences are anticipated due to the reversibility of a regular carbamate, which is eliminated when substituted with a methyl group.³⁹ Indeed, when analyzing the temperatures where a 5% mass loss occurs ($T_{d\ 5\%}$), clear differences between **H₈-ene** and **M₈-ene** are observed, indicating that methylation increases the thermal stability by 50 °C. However, this trend is not seen in the OH-terminated oligomers ($T_{d\ 5\%}$ of **H₈-OH** is 247 °C and $T_{d\ 5\%}$ of **M₈-OH** is 244 °C, see Fig. S28†), suggesting that also the terminal groups influence the thermal stability. Previous research studies have demonstrated that

these OH-endcapped oligourethanes have a self-immolative character, undergoing a 5-*exo-trig* cyclization initiated by an alkaline trigger.^{17,37,40} Although no basic conditions are present, it is possible that the cyclization occurs under thermal conditions, which could explain the earlier observed absence of differences in thermal stability. Indeed, comparing the thermal stability of the ene-terminated oligomers with the corresponding OH-terminated alternatives, a clear trend can be observed. Across all oligomers, the $T_{d\ 5\%}$ values improve when the potential for backbiting is eliminated, indicating the cyclization as a predominant degradation pathway.

To validate this assumption, pyrolysis-GC-MS analysis of the ene-terminated oligomers was conducted above their respective degradation temperatures (see Tables S1–S4†). While the presence of oxazolidin-2-one can be attributed to both a dissociative intramolecular transcarbamoylation and a cyclization reaction, 3-methyl-2-oxazolidinone can only be generated through the latter. Because this is the main degradation product in **M₈-ene**, **B₈-ene** and **A₈-ene** (see Tables S1–S4†), this serves as additional proof that the cyclization is indeed the predominant degradation pathway. Additionally, this implies that the thermal stability is dictated by the generation of the terminal alcohol, which is determined by the integrity of the (terminal) carbamate moieties. This translates into a substantial enhancement in the thermal stability of the **M₈-ene** oligomer relative to the **H₈-ene**, **B₈-ene**, and **A₈-ene** oligomers (see Fig. 4b), given the increased resilience of Me-substituted carbamate. Allyl isocyanate detected in the pyrolysis-GC-MS spectrum of **H₈-ene**, **B₈-ene**, and **A₈-ene** provides additional confirmation of the reversibility and resulting lability of the N–H carbamate bonds in these oligomers. These findings offer valuable insights into the thermostability and degradation pathway of oligourethanes, enabling precise material design through careful selection of the design parameters.

Conclusion

First, we demonstrated that sequence-defined oligourethanes can be produced using a straightforward and scalable protocol, allowing the preparation of up to 50 g of these uniform oligomers on lab scale. Through a bidirectional process, we

Table 2 Monoisotopic masses and thermal properties of the investigated octamers

Sequence	m/z_{th}^a	m/z_{exp}^b	Yield ^c (%)	T_g^d (°C)	T_m^e (°C)	$T_{d\ 5\%}^f$ (°C)	ΔH_{fus}^g (kJ mol ⁻¹)
M₈-OH	983.5507	983.5529	80.6	-24	—	244	—
M₈-ene	975.5609	975.5583	77.6	-34	—	302	—
H₈-OH	871.4255	871.4235	88.4	—	150	247	151
H₈-ene	863.4357	863.4333	85.7	—	158/168	250	159
B₈-OH	927.4881	927.4886	82.8	-42	—	220	—
B₈-ene	919.4983	919.4974	78.1	-13	—	245	—
A₈-OH	927.4881	927.4885	75.2	-26	—	216	—
A₈-ene	919.4983	919.4971	86.3	-11	70	249	58

^aTheoretical monoisotopic m/z . ^bDetermined monoisotopic m/z via orbitrap-MS measurements. ^cYields obtained for the entire synthesis process. ^d T_g determined via the second heating run in DSC. ^eMelting temperature determined via the first heating run in DSC. ^fThermal onset-temperature of 5% weight-loss measured via TGA. ^gEnthalpy of fusion integrated from the first heating run in DSC.



reduced the synthetic steps and achieved precise control over the length, microstructure, and end-functionality of the targeted macromolecules. Hence, non-substituted, substituted, blocky, and alternating octamers bearing OH- or alkene-terminated groups have been synthesized. Subsequent thermal analysis gave important insights into the parameters that dictate the thermal transitions characterizing these products, including T_g , melting temperatures and melting enthalpies. This research was specifically focused on the effect of hydrogen bonding, which is present in N-H carbamates but absent in the methyl-substituted monomers. While increasing the amount of these non-covalent interactions showed an increase in both melting temperature and enthalpy, the chain length could be used as a parameter to regulate the T_g of this type of materials. Furthermore, the arrangement of the monomers plays a crucial role in shaping these thermal properties, as illustrated by the fact that **B₈-ene** is amorphous while **A₈-ene** is a semi-crystalline structure. Thermal degradation was also examined, revealing no significant differences among the OH-terminated oligomers, which contradicts the expectation that methylated carbamates would be more stable due to their lack of reversibility. However, when the alcohol end group was replaced with an allyl group, the anticipated behavior emerged, showing that conventional urethanes exhibited inferior thermal stability. This indicated that cyclization, rather than carbamate breakage, is the primary degradation pathway. This conclusion was supported by pyrolysis-GC-MS measurements. The information gathered in this work offers valuable insights into the structure–property relationships of PUs, emphasizing design parameters that allow their tailor-made engineering and thus paving the way to the development of next-generation polyurethane materials.

Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Data availability

The data supporting this article have been included as part of the ESI.†

Conflicts of interest

The authors declare no competing financial interest.

Acknowledgements

This project received funding from the European Research Council (ERC) under the European Union's Horizon 2020 Research and Innovation Programme 101021081

(ERC-AdG-2020, CiMaC-project). J. V. H. acknowledges the Research Foundation-Flanders (FWO) for his Ph.D. fellowship under application number 11PJ624N. The NMR expertise centre (Ghent University) is also acknowledged for providing support and access to NMR infrastructure funded by a Research Foundation Flanders (FWO I006920N) and the Bijzonder Onderzoeksfonds (BOF.BAS. 2022.0023.01). Tine Van Laere, Dr. Dave Manhaeghe and Professor Steven De Meester are acknowledged for the pyrolysis GC-MS measurements. Pieter Surmont is thanked for the orbitrap-MS measurements and Bernhard De Meyer for assistance regarding thermal analyses. The authors thank Dr. Susanne Fischer for the experimental help and both Dr. Matthieu Soete and Dr. Susanne Fischer for the fruitful discussions.

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