

## REVIEW

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Hydrogels with electrically conductive  
nanomaterials for biomedical applications†

Georgios Kougkoulos,<sup>ab</sup> Muriel Golzio,<sup>c</sup> Lionel Laudebat,<sup>bd</sup>  
Zarel Valdez-Nava<sup>ab</sup> and Emmanuel Flahaut<sup>ab</sup>

Hydrogels, soft 3D materials of cross-linked hydrophilic polymer chains with a high water content, have found numerous applications in biomedicine because of their similarity to native tissue, biocompatibility and tuneable properties. In general, hydrogels are poor conductors of electric current, due to the insulating nature of commonly-used hydrophilic polymer chains. A number of biomedical applications require or benefit from an increased electrical conductivity. These include hydrogels used as scaffolds for tissue engineering of electroactive cells, as strain-sensitive sensors and as platforms for controlled drug delivery. The incorporation of conductive nanomaterials in hydrogels results in nanocomposite materials which combine electrical conductivity with the soft nature, flexibility and high water content of hydrogels. Here, we review the state of the art of such materials, describing the theories of current conduction in nanocomposite hydrogels, outlining their limitations and highlighting methods for improving their electrical conductivity.

## 1. Introduction

## 1.1 Hydrogels

Hydrogels are three-dimensional, viscoelastic networks of hydrophilic polymer chains, cross-linked in an aqueous environment. The water-polymer network is gelled through the association of polymer chains to form a continuous structure, that immobilises water within it and becomes resistant to

<sup>a</sup> CIRIMAT, Université de Toulouse, CNRS, INPT, UPS, 31062 Toulouse CEDEX 9, France. E-mail: [emmanuel.flahaut@univ-tlse3.fr](mailto:emmanuel.flahaut@univ-tlse3.fr)

<sup>b</sup> LAPLACE, Université de Toulouse, CNRS, INPT, UPS, 31062 Toulouse CEDEX 9, France. E-mail: [zarel.valdez-nava@laplace.univ-tlse.fr](mailto:zarel.valdez-nava@laplace.univ-tlse.fr)

<sup>c</sup> IPBS, Université de Toulouse, NRS UMR, UPS, 31077 Toulouse CEDEX 4, France

<sup>d</sup> INU Champollion, Université de Toulouse, 81012 Albi, France

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From left to right: Georgios Kougkoulos, Muriel Golzio, Lionel Laudebat, Zarel Valdez-Nava and Emmanuel Flahaut

Emmanuel Flahaut is a senior CNRS researcher in materials science. His research is focused on the synthesis and functionalization of carbon nanomaterials as well as their application in various fields ranging from materials science to biomedicine. All are affiliated to the University of Toulouse, France. They are all partners of a National (ANR) project on non-invasive drug delivery through skin electroporation, involving electrically conductive nanocomposite hydrogels.

Georgios Kougkoulos is pursuing a PhD in materials science. His research interests lie in the interface between materials science and biomedicine and currently include nanocomposite hydrogels, drug delivery and electroporation.

Muriel Golzio is a senior CNRS researcher in biophysics. Her main research interests lie in the field of cell electroporation, nucleic acids electrotransfer and biomedical applications for cancer treatment and gene therapy.

Lionel Laudebat is an associate professor of electrical engineering and researcher. His main research interests include numerical simulation and modelling of power electronic materials and studies of dielectric materials. Zarel Valdez-Nava is a CNRS researcher in electrical engineering. His research concerns insulating and dielectric materials for high voltage and/or high temperature power electronics, focusing in identifying the relationship between processing and properties of composites and ceramics.

flow.<sup>1</sup> The association of polymer chains occurs through a variety of mechanisms that can be classified into physical or chemical. Mechanisms of physical cross-linking include the physical entanglement of individual polymer chains, typically induced by solubility alterations in response to temperature changes; macromolecular self-assembly through non-covalent bonding (hydrogen bonds, van der Waals forces, hydrophobic interactions); crystallization, the formation of microcrystals through a freeze-and-thaw process, which then act as a cross-linking site; ionic gelation, polymer chains surround ions forming a crosslinking site; and electrostatic interaction of polymer chains with opposite charges.<sup>2–5</sup> Chemical cross-linking involves the formation of covalent bonds between polymer chains through mechanisms including radical polymerization, chemical reactions of complementary groups (for example hydroxyl groups or amides with carboxylic acids), high energy irradiation, addition of chemical cross-linking agents (glutaraldehyde, epoxy compounds, isocyanates, metal ions) and enzymatic reactions.<sup>2–4,6</sup>

Physically cross-linked hydrogels are reversible, easy to produce and do not require the use of chemical cross-linking agents.<sup>3,4</sup> Chemically cross-linked hydrogels generally have higher mechanical strength and offer more possibilities for control of the cross-linking process and customised design.<sup>2,3</sup> However, they often make use of toxic cross-linkers which then have to be extracted to keep the hydrogel biocompatible.<sup>4</sup> The combination of physical and chemical cross-linking mechanisms offers the possibility for more precise control over the hydrogels' properties.<sup>2</sup>

Hydrogels can be further classified according to their polymeric composition as: (1) single polymer networks (homo-polymers), derived from one type of monomer unit; (2) copolymers, derived from the cross-linking of two or more types of monomers, arranged in alternating, block or random configuration on the polymer chain; (3) semi-interpenetrating polymer networks, where a linear polymer is contained within an independent, cross-linked polymer network; and (4) interpenetrating polymer networks, derived from two independent, cross-linked polymer networks interlocked together.<sup>4,5,7</sup> In each case, at least one of the monomers must be hydrophilic, to render the network water swellable.

Hydrogels are rich in water (typically 80 to 95% w/w but can span almost all range), flexible with viscoelastic behaviour and usually biocompatible.<sup>8–10</sup> These properties have fostered numerous applications in the domains of biomedicine, soft electronics and actuators.<sup>2</sup> Some of these properties can be straightforwardly modified to fit a specific application. For example, the rigidity and water content of hydrogels can be tuned by adjusting the polymer concentration and the cross-linking degree.<sup>11</sup> In an aqueous environment, a dynamic equilibrium exists between the interactions responsible for water sorption (capillary, osmotic and hydration forces) and the cross-linked polymer network resisting expansion.<sup>12</sup>

In addition, hydrogels can be highly responsive to external stimuli. Small changes in environmental conditions such as temperature, pH, pressure, electric field or chemical agents can induce unexpected and mostly reversible changes in hydrogel properties such as volume, swelling degree, conductivity or

permeability.<sup>13</sup> These stimuli-responsive or smart hydrogels can be engineered to sense external stimuli and transmit an electrical or optical signal, by modifying a readily readable property such as electrical conductivity or colour.<sup>13,14</sup> Many hydrogels used in motion sensing devices change their electrical conductivity when strained, giving an electrical signal to the device.<sup>13</sup> Hydrogel membranes can selectively allow the transfer of molecules by swelling to open or block their pores, according to biochemical signals.<sup>15</sup> More recently, 3D printing of hydrogels has opened new possibilities for customized design.<sup>16</sup>

Despite their remarkable properties and several possible applications, hydrogels have some important limitations. They generally have limited mechanical strength and are susceptible to irreversible deformation.<sup>17</sup> In addition, hydrogels from commonly used polymers intrinsically have very low electrical conductivity. These limitations make conventional hydrogels unsuitable for applications that require robustness or electrical conductivity.

## 1.2 Electrically conductive hydrogels

Electrical conductivity, in particular, is an important property of hydrogels in biomedical applications. Applications of electrically conductive hydrogels include hydrogels used as substrates for the growth of electroactive cells, hydrogels which function as flexible strain sensors for health monitoring and wearable devices and drug-loaded hydrogels used for electro-stimulated drug delivery.<sup>18</sup> There are three approaches to prepare electrically conductive hydrogels: (1) using a conducting polymer in the hydrogel matrix, (2) increasing the ionic conductivity of the aqueous phase and (3) incorporating a conductive material in the hydrogel.

Conducting polymers are organic macromolecules with intrinsic electrical conductivity. Conducting polymers commonly used for preparing hydrogels are polypyrrole (PPy), polyaniline (PANI) and poly-(3,4-ethylenedioxythiophene) (PEDOT).<sup>19</sup> They cannot form hydrogels themselves but have to be combined with a supporting polymer that provides hydrophilicity and mechanical strength.<sup>19</sup> They provide a conducting path of electronic conductivity due to the delocalized pi electrons of their conjugated systems, and can also increase the ionic conductivity of the aqueous phase by contributing ions.<sup>20,21</sup> Guo and Ma reviewed materials for tissue engineering with conducting polymers, including hydrogels<sup>22</sup> and Stejskal reviewed conducting polymer hydrogels with a focus on preparation methods.<sup>19</sup>

The ionic conductivity of the aqueous phase can be increased by preparing and/or swelling the hydrogel in an ion-rich aqueous solution. Free ions can be generated in water from acids, metal salts or ionic liquids.<sup>23</sup> Metal ions, in particular, can act both as cross-linking agents for polymer chains and electrolytes of ionic conductivity.<sup>24</sup> The cross-linking mechanism is metal-ligand interaction where the metal ions form coordinate covalent bonds with chelating agents (N, O, S) in the polymer chain.<sup>25</sup> In high concentrations, the free metal ions contribute to the ionic conductivity of the hydrogel.<sup>24</sup> Zhang *et al.* review the recent advances in metal ion hydrogels for biological applications.<sup>25</sup> Non-ionic polymer chains may impair the ionic conductivity of



hydrogels, compared to polyelectrolyte hydrogels.<sup>26</sup> Polyelectrolytes are polymers which contain ionic and/or ionizable groups in a substantial portion of their constitutional units.<sup>27</sup> They can be cationic, anionic or ampholytic (containing both negative and positive charges; also called zwitterionic, commonly when the positive and negative charges are located in the same pendant group).<sup>26,27</sup> In low concentration electrolyte solutions, polycationic and polyanionic hydrogels exhibit the highest ionic conductivity, due to high concentrations of mobile counterions. In high concentration electrolyte solutions, polyzwitterionic hydrogels promote fast ion dissociation and transport due to the highly charged polar side groups.<sup>26,28</sup> Wang *et al.* and Liu *et al.* reviewed polyelectrolyte<sup>29</sup> and polyzwitterionic<sup>30</sup> hydrogels for biomedical applications.

The incorporation of conductive materials aims to create a continuous network of electronic conduction throughout the hydrogel, through the dispersion of conductive nanomaterials and/or other conductive materials, such as metal microwires or carbon fibers, in the hydrogel framework.

The current review focuses on the incorporation of conductive nanomaterials in otherwise electrically insulating hydrogel polymer matrices. Each section is accompanied by illustrative examples of practical applications from the recent literature. Electrically conductive hydrogels are widely researched for applications in biomedicine and soft electronics including tissue engineering, strain sensors and controlled drug release (Fig. 1). In the following section, we present these practical applications and underline the relevance of electrical conductivity in each case. Section 2 lists the conductive nanomaterials commonly incorporated in hydrogels. In Sections 3–5 we delve into the electrical properties of nanocomposite hydrogels by

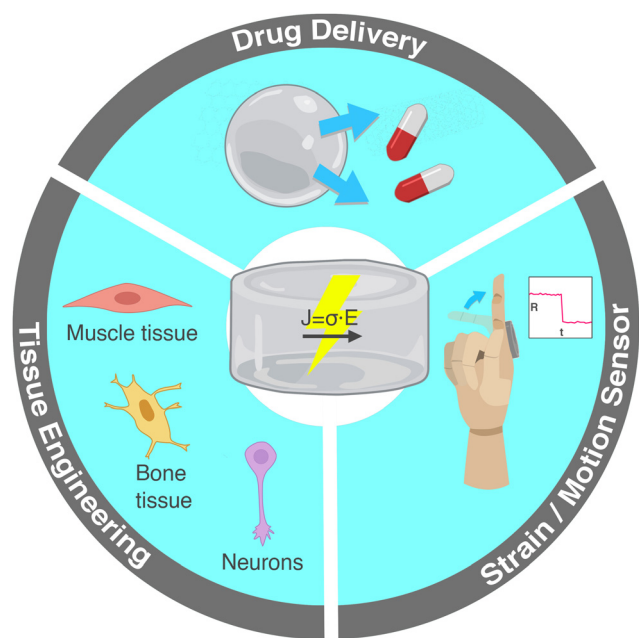
introducing electrical percolation, current conduction mechanisms and experimental electrical characterization techniques. Lastly, section 6 includes a comprehensive table and graph of conductive nanocomposite hydrogels and a discussion on some processing methods that can improve electrical conductivity.

### 1.3 Applications

**1.3.1 Tissue engineering.** Tissue engineering aims to develop artificial tissue and organs to restore functions of injured or malfunctioning organs *in vivo*, by cultivating cells on suitable substrates.<sup>31</sup> Hydrogels are particularly adequate as substrates because they can be biocompatible and biodegradable, have a similar water content and stiffness as natural organs and their porosity allows for sufficient nutrient transfer to the growing cells (Fig. 2).<sup>32,33</sup> The substrate is responsible for cell adhesion, proliferation and differentiation, functions physiologically performed by the extracellular matrix, the scaffold that provides structural and biochemical support to surrounding cells.<sup>34</sup>

Electrical conduction plays an important role in cell signalling and affects the differentiation and proliferation of cells. In particular, muscle and neural cells are excitable; they use rapid changes in membrane potential for cell-to-cell communication. Muscle tissue cells can contract, in response to electrical signals. Thus, substrates with an electrical conductivity near the conductivity of the native muscle deliver the most promising results.<sup>35</sup>

Navaei *et al.* developed a conductive, nanocomposite hydrogel to act as a cardiac tissue substrate for myocardial regeneration and repair. They synthesized gold nanorods which were incorporated into a methacrylated gelatin prepolymer solution and cross-linked it through UV radiation. A concentration of  $1.5 \text{ mg ml}^{-1}$  of gold nanorods reduced the electrical impedance of the hydrogel (measured by impedance spectroscopy) and improved both cell–cell signalling and electrical signal propagation on cells growing on the substrate, compared to the pristine hydrogel. This resulted in an improved cardiac tissue contractility with a lower excitation threshold (Fig. 3(A)).<sup>36</sup> Dong *et al.* reviewed the conductive biomaterials employed in muscle tissue engineering<sup>35</sup> while a review article of Li *et al.*



**Fig. 1** Main applications of hydrogels with electrically conductive nanomaterials: tissue engineering of electroactive cells, controlled drug delivery and strain sensors for human motion monitoring.



**Fig. 2** Biodegradable hydrogel as a substrate for tissue regeneration. (A) Stem cells seeded on hydrogel. (B) Hydrogel degrades while cells proliferate and secrete their extracellular matrix. (C) Hydrogel has completely degraded leaving in place a new tissue composed of tissue and extracellular matrix. Reproduced from ref. 32 with permission from The Royal Society of Chemistry.





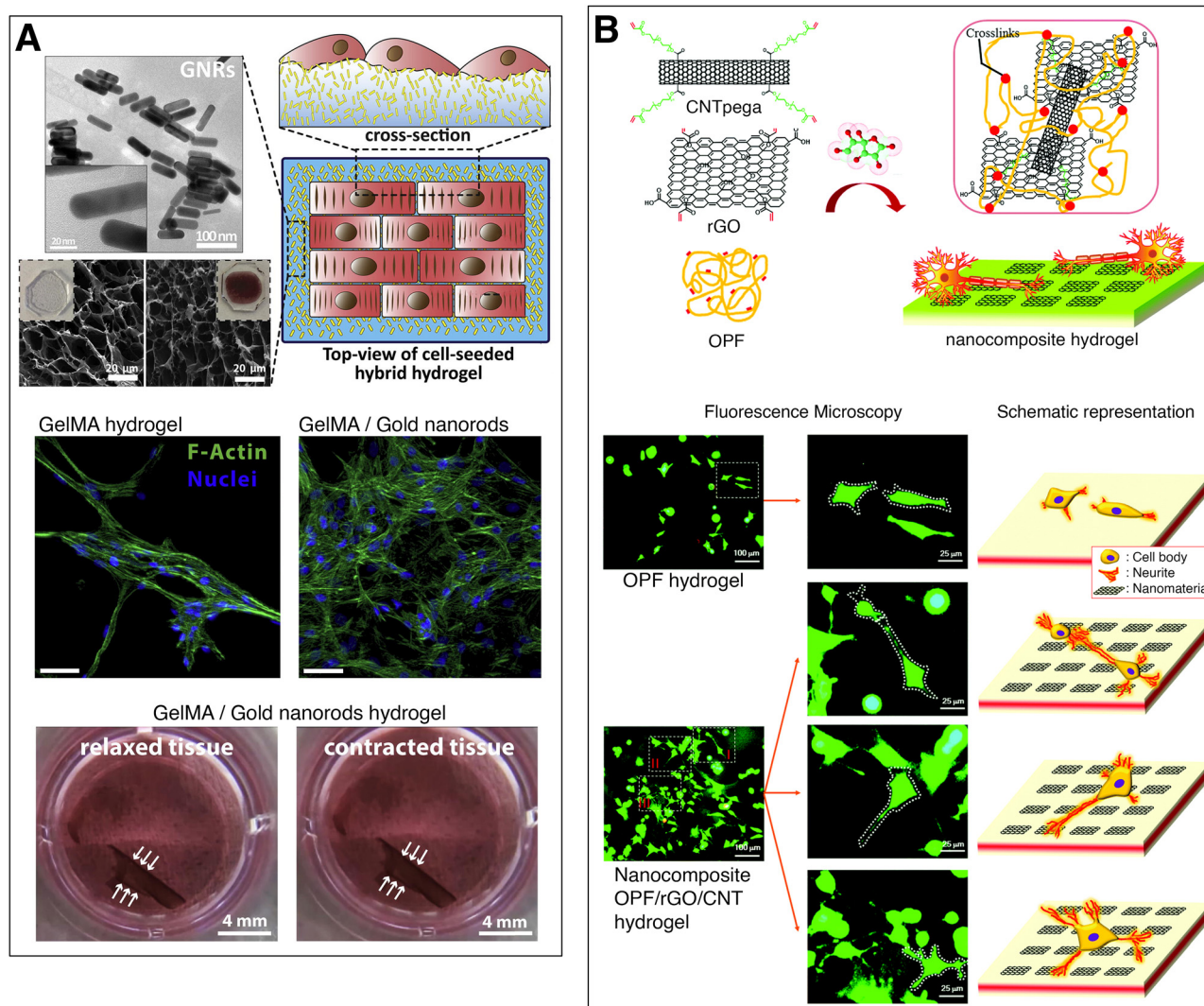
focused on the conductive biomaterials employed for cardiac repair.<sup>37</sup>

Nerve tissue is comprised of electroactive neural cells that transmit electrochemical signals. Conductive scaffolds, enhance the cellular activity and tissue regeneration of neural cells, compared to non-conductive substrates.<sup>38</sup> Liu *et al.* chemically cross-linked reduced graphene oxide (rGO) sheets and carbon nanotube (CNT) poly(ethylene glycol) acrylate to oligo(poly ethylene glycol) fumarate hydrogels, increasing their electrical conductivity from  $2 \times 10^{-4} \text{ S m}^{-1}$  to  $7.9 \times 10^{-4} \text{ S m}^{-1}$ . The nanoengineered conductive hydrogel was biocompatible and led to a higher proliferation of PC12 cells and increased neurite development (Fig. 3(B)).<sup>39</sup>

Bone regeneration and remodelling involves electrical fields generated by intrinsic piezoelectric properties of the bone tissue.<sup>33</sup>

Additionally, bone cell tissue engineering requires scaffolds with high mechanical strength to stimulate the differentiation of osteocytes.<sup>40</sup> Pelto *et al.* fabricated polylactide scaffolds to serve as a substrate for osteogenic differentiation and coated them with electrically conductive polypyrrole. The electrically conductive scaffold increased significantly the differentiation of human adipose stem cells compared to the insulating one.<sup>41</sup>

**1.3.2 Strain sensors.** Flexible and soft strain sensors are used in wearable devices that detect motion or pressure. This is relevant in the fields of health monitoring, prosthetics, soft robotics, electronic skin and human-machine communication.<sup>42</sup> Stimuli-responsive hydrogels are ideal materials for these applications because of their flexibility, stretchability, self-healing ability and biocompatibility.<sup>13,43</sup> Electrical conductivity is a readily measurable property of hydrogels and can be directly translated to electronic

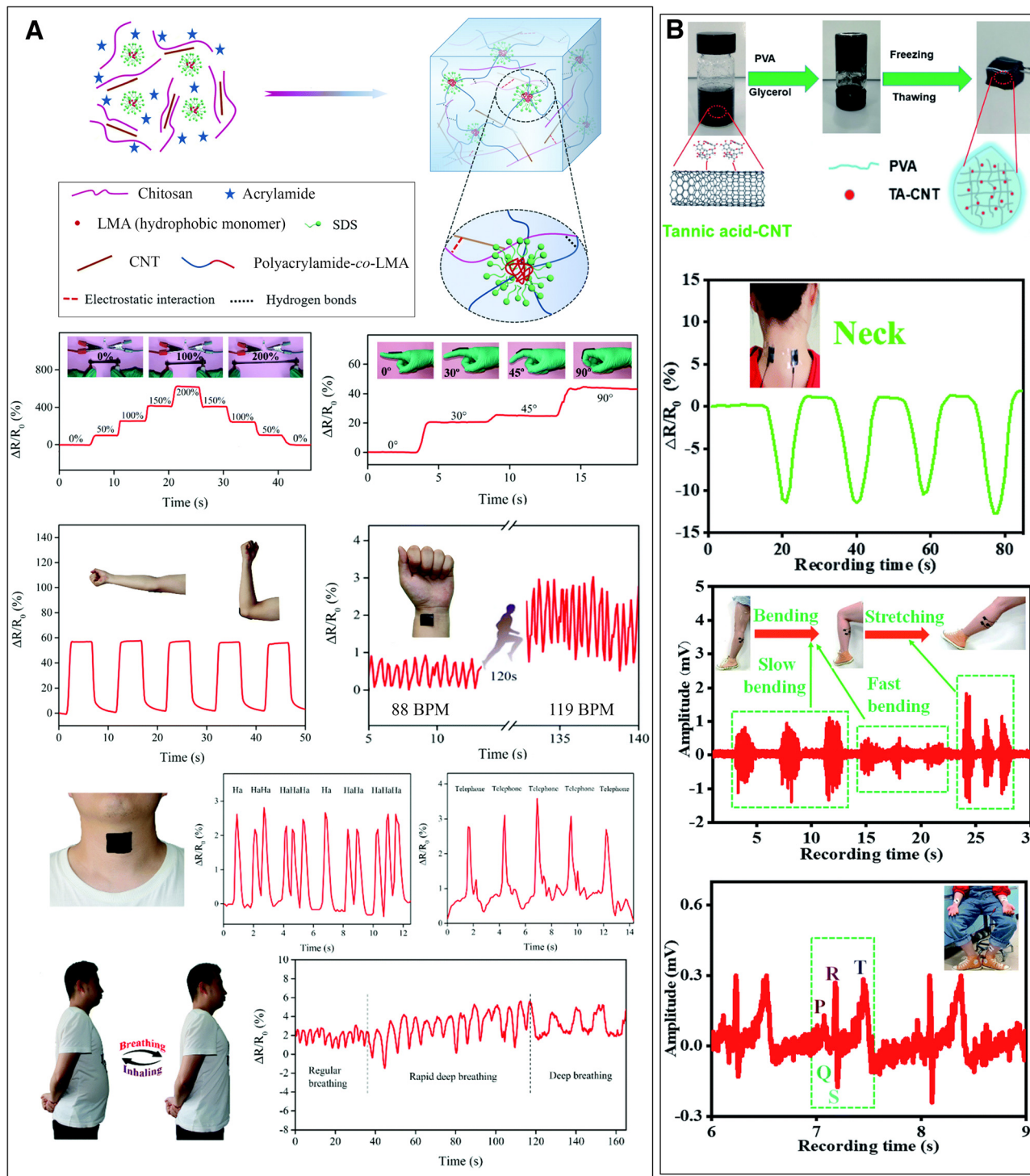


**Fig. 3** Electrically conducting hydrogels for tissue engineering. (A) Gelatin methacrylate hydrogel with gold nanorods for myocardial regeneration. The nanocomposite hydrogels illustrate highly packed cardiomyocytes with local alignment of fibers. A cell-loaded nanocomposite hydrogel displays contraction of the full hydrogel. Adapted from ref. 36, Copyright 2016, with permission from Elsevier. (B) Oligo(poly ethylene glycol) fumarate hydrogel with CNTs and rGO for neural tissue engineering. Fluorescent microscopy images and schemes of PC12 cells growing on plain and nanocomposite hydrogel, showing that neural cells growing on conductive substrates exhibit increased neurite development. Adapted from ref. 39 with permission from The Royal Society of Chemistry.



wearable, self-healable and adhesive sensor (Fig. 4(A)).<sup>44</sup> The incorporation of 1% w/w CNTs increased the conductivity of the hydrogel from 0.1 to 0.95 S m<sup>-1</sup> and more importantly improved the strain sensitivity of the hydrogel (the ratio of

**Fig. 4** Conductive nanocomposite hydrogels used as strain sensors for monitoring human motions and health. (A) A polyacrylamide-co-lauryl methacrylate/chitosan and carboxylated CNTs hydrogel changes its electrical resistance with strain. This property can be exploited to monitor movements such as finger and elbow flexion or even heart pulse, talking and breathing. Adapted from ref. 44 with permission from The Royal Society of Chemistry. (B) Polyvinyl alcohol/tannic acid–CNT hydrogel monitors joint movements such as neck bending and can also be used as an electrode for electrophysiological measurements (EMG and ECG). Adapted from ref. 46 with permission from The Royal Society of Chemistry.



**Fig. 4** Conductive nanocomposite hydrogels used as strain sensors for monitoring human motions and health. (A) A polyacrylamide-co-lauryl methacrylate/chitosan and carboxylated CNTs hydrogel changes its electrical resistance with strain. This property can be exploited to monitor movements such as finger and elbow flexion or even heart pulse, talking and breathing. Adapted from ref. 44 with permission from The Royal Society of Chemistry. (B) Polyvinyl alcohol/tannic acid–CNT hydrogel monitors joint movements such as neck bending and can also be used as an electrode for electrophysiological measurements (EMG and ECG). Adapted from ref. 46 with permission from The Royal Society of Chemistry.

resistance change over applied strain). The electrical resistance of the hydrogel increased with increasing stress and recovered fast (150 ms) after the stress was removed.<sup>44</sup> The strained hydrogel both reorganized the conductive nanomaterial network decreasing its electronic conductivity and densified the polymer network, restricting ion mobility and thus reducing ionic conductivity.<sup>44,45</sup> The hydrogel was tested for the monitoring of movements of human articulations (finger, wrist, elbow, knee) and was sensitive enough to monitor even subtle strains such as talking, breathing and pulse, placed on the neck, chest and inside of wrist, respectively (Fig. 4(A)).<sup>44</sup> He *et al.* developed a highly-sensitive, freeze-resisting strain sensor gel by incorporating tannic acid-CNTs into a polyvinyl alcohol water/glycerol gel. The nanomaterials increased the electrical conductivity from 0.19 to 5.13 S m<sup>-1</sup> and the gels could be used as strain sensors to detect joint movements and as flexible electrodes for the detection of electrophysiological signals (EMG and ECG, Fig. 4(B)).<sup>46</sup> Coupled with a wireless transmitter, such hydrogel materials can be useful in health and activity monitoring.<sup>45</sup> Three recent review publications summarised the latest developments regarding hydrogels used as strain sensors.<sup>13,42,43</sup>

Nanocomposite hydrogels used as biosensors are the ideal candidates to bridge the gap between humans and machines, in the context of biomedical engineering, diagnostics and wearable and implantable devices.<sup>47</sup> Both living organisms and electronic devices use electric signals to coordinate their activities. In living organisms, these signals are based on ionic conduction in a soft and water-rich environment. Small ions (Na<sup>+</sup>, K<sup>+</sup>, Cl<sup>-</sup>, Ca<sup>2+</sup>) rapidly flow through channels in the plasma membranes of cells, hyperpolarising and depolarising the transmembrane potential. In machines, electric signals are based on electronic conductivity in dry and rigid solids. Free electrons flow through the metallic or semi-conducting components of electronic devices. Nanocomposite hydrogels incorporate elements of both: they are viscoelastic materials with adjustable mechanical strength, high water content and solid-like behaviour, and the conduction mechanisms can combine ionic and electronic conductivity. Yuk *et al.* have reviewed conductive hydrogels in the interface between biology and electronics.<sup>47</sup>

**1.3.3 Drug delivery.** Conventional drug delivery has a number of drawbacks including high dosages, limited bioavailability, repeated administration and potential toxicity.<sup>48</sup> Controlling how, when and where drugs are available to cells and tissues can increase the drug's efficiency and reduce the frequency and concentration of the doses, limiting the toxicity and improving patient compliance and life quality. Hydrogels are some of the most promising and widely considered platforms for controlled drug delivery. They are hydrophilic, biocompatible, have a large water retention capacity and can be stimuli-responsive. Typical mesh sizes of hydrogels range from 10 to 100 nm.<sup>48</sup> This means that most molecules can diffuse freely within the porosity of the hydrogel network. The mesh size can be decreased by increasing the polymer concentration and cross-linking, to delay the release rate by steric hindrance.

To further reduce the release rate, active ingredients can form covalent, electrostatic or hydrophobic interactions with the polymer matrix.<sup>48</sup> Nanomaterials can also reduce the release rate of active ingredients.<sup>48,49</sup> Hydrophilic drugs can be readily diffused within hydrogels while hydrophobic drugs can associate with hydrophobic domains (aliphatic chains, cyclodextrin) in the polymer network or be encapsulated in nanovesicles.<sup>48,50</sup> Hydrogels can be engineered to release part of their water content according to environmental stimuli.<sup>51</sup> Pulsatile drug release, in particular, can mimic the natural patterns of *in vivo* release of endogenous chemicals such as insulin, growth hormone and oestrogen.<sup>48,52</sup> Externally applied electric fields can be used to control the release rate of drugs from conductive hydrogels. Servant *et al.* added ball-milled graphene nanosheets to methacrylic acid hydrogels, to improve their mechanical and electrical properties. A concentration of 0.2 mg ml<sup>-1</sup> of graphene nanosheets increased the electrical conductivity of the hydrogel from  $2.9 \times 10^{-6}$  to  $10^{-5}$  S m<sup>-1</sup>. The nanocomposite conductive hydrogels demonstrated controlled, pulsatile release of a small molecule (sucrose) upon the intermittent application of an electrical field (Fig. 5(A)).<sup>53</sup> Additionally, nanomaterials can enhance electro-stimulated drug release. Liu *et al.* incorporated rGO into poly(vinyl alcohol) hydrogels and loaded them with a drug (lidocaine). With no external stimulation, the rGO nanomaterials acted as a barrier, retaining the drug within the hydrogel, while the application of an electric field triggered the release of the drug. The addition of rGO negatively charged the polymer matrix, enhancing electro-osmosis. In contrast, a control hydrogel without conductive nanomaterials did not change the drug release profile with electric field application (Fig. 5(B)).<sup>49</sup> Merino *et al.* reviewed the field of nanocomposite hydrogels for controlled drug delivery.<sup>51</sup>

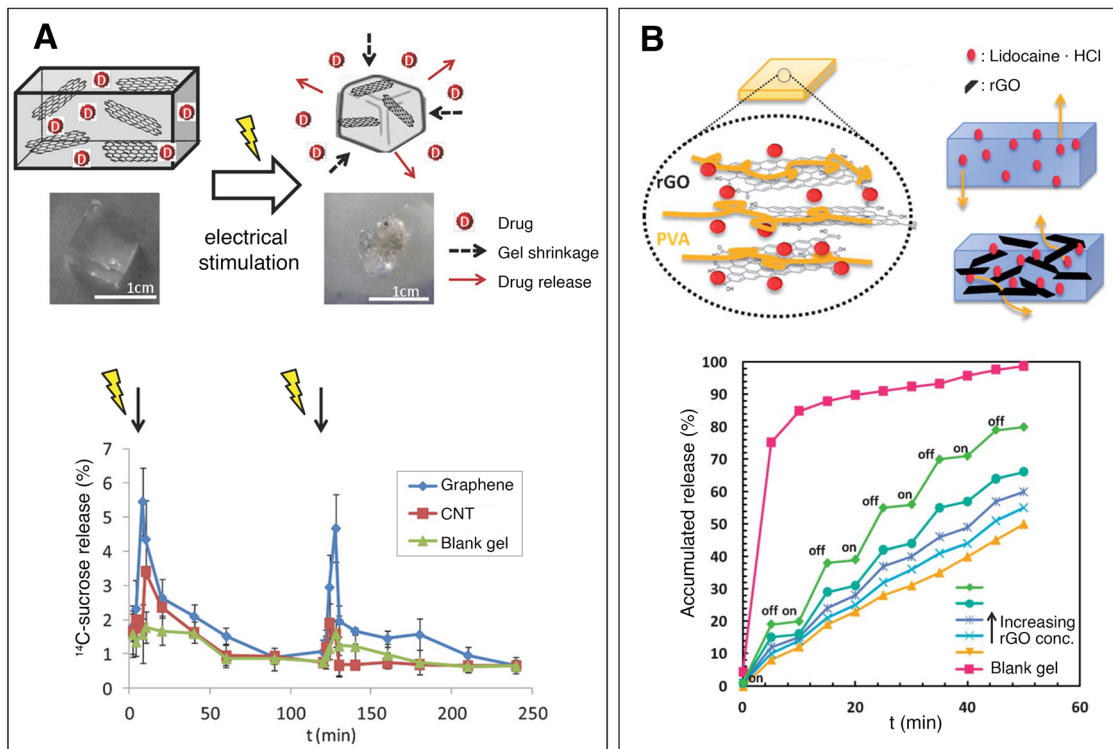
## 1.4 Electrical conductivity targets

**1.4.1 Tissue engineering.** The electrical conductivity values aimed through the incorporation of conductive nanomaterials depend directly on the application. Hydrogels used as substrates for tissue engineering of electroactive cells require conductivity values close to native tissues. The conductivity values for tissues depend on the electrical measurement method, investigated species and if it was measured *in vivo* or *ex vivo*, and have been reported to range from 0.15 to 2.6 S m<sup>-1</sup> for muscle tissue,<sup>35,54</sup> 0.4 to 3 S m<sup>-1</sup> for nerves,<sup>54,55</sup> for bones the values range from  $9 \times 10^{-3}$  S m<sup>-1</sup> (cortical bone) to 0.23 S m<sup>-1</sup> (bone marrow).<sup>56</sup> The electrical environment of cells plays an important role in the development of healthy tissues. In addition, conductive hydrogels used as substrates for tissue engineering can serve as electrodes for the application of external electrical stimulation to cells. Electric fields and currents affect tissue and cellular behaviour in a number of ways including interference in differentiation, migration, alignment, cytoskeleton organization, neurite growth in neurons, calcification of osteoblasts, collagen production from fibroblasts and wound healing.<sup>57–59</sup>

**1.4.2 Strain sensors.** Biosensors based on conductive hydrogels function by measuring a resistance change and correlating this change to a signal. The signal can be a motion,







**Fig. 5** Conductive nanocomposite hydrogels for controlled drug delivery. (A) Electrical stimulation causes a methacrylic acid–graphene nanosheet hydrogel to shrink, releasing drug. *In vivo* release profile of  $^{14}\text{C}$  sucrose on the blood plasma of mice implanted with sucrose-loaded hydrogels. Hydrogels with no nanomaterials (green),  $0.2\text{ mg mL}^{-1}$  CNTs (red) and  $0.2\text{ mg mL}^{-1}$  graphene (blue) were tested. A tension of  $10\text{ V DC}$  is applied for  $1\text{ minute}$  with a time interval of  $2\text{ h}$ . The graphene-loaded hydrogel (which also had the highest conductivity) demonstrated a pulsatile release of sucrose, controlled by the electric field. Adapted, with permission, from ref. 53. Copyright 2014 Wiley. (B) Reduced graphene oxide nanomaterials retain lidocaine hydrochloride within a polyvinyl alcohol hydrogel. Upon pulsatile electrical stimulation (on/off,  $15\text{ V DC}$ ) the nanocomposite hydrogels exhibit controlled drug release, while the blank gel does not change release profile. Adapted from ref. 49 with permission from The Royal Society of Chemistry.

as in strain sensors for health monitoring (joints movements, breath *etc.*) or a biosignal, such as the attachment of an antigen on the hydrogel.<sup>45,60</sup> The mechanism through which the resistance of a conductive nanocomposite hydrogel changes with strain is triple: firstly, a strain applied on the hydrogel changes the hydrogel's geometry, affecting the overall resistance. Secondly, the polymer matrix densifies (or loosens), affecting ion mobility. Thirdly, a strain changes the configuration of the conductive nanomaterial network within the hydrogel, breaching or creating electrical pathways. For this application, a hydrogel with a higher conductivity ensures a more reliable signal. The nanomaterial concentration ideally has to be maintained slightly higher than the percolation concentration.<sup>‡</sup> A very high nanomaterial concentration risks to “saturate” the electrical network, resulting in minor conductivity changes under strain: a highly percolating network will remain percolated even with some elongation.

**1.4.3 Drug delivery.** In nanocomposite hydrogels for drug delivery, two approaches are distinguished: electro-responsive hydrogels that release a drug upon application of an external electric field and conductive hydrogels used as electrodes for transdermal drug delivery through skin electroporation.<sup>51,61</sup> Electro-responsive hydrogels shrink or bend upon the application

of an external electric field. A loaded drug is released through the contraction of the hydrogel as well as through electrophoretic forces. The electrically-induced contraction of hydrogels occurs through the combination of four mechanisms: (1) a stress gradient in the hydrogel, (2) electro-osmosis of water coupled with electrophoresis, (3) local pH changes near the electrodes due to water electrolysis and (4) a temperature gradient in the hydrogel due to resistive heating.<sup>51</sup> So far, there is no clear conductivity target for electro-responsive hydrogels in controlled drug release, however, the incorporation of conductive nanomaterials has been shown to enhance drug release.<sup>49,53</sup> In the case of conductive hydrogels for transdermal drug delivery, the hydrogel functions as an electrode, for the application of pulsed electric field on the skin.<sup>61</sup> A higher conductivity ensures a more efficient distribution of the electric field into the skin, a prerequisite for skin electroporation.<sup>61,62</sup> According to FEM simulations run on a skin model, a hydrogel conductivity higher than  $10^{-4}\text{ S m}^{-1}$  ensures a critical distribution of the pulsed electric field in the skin, for a side-to-side electrode configuration and  $300\text{ V}$  potential difference.<sup>63</sup>

## 2. Conductive nanomaterials

The conductive nanomaterials incorporated in hydrogels to increase their electrical conductivity fall into three categories:

<sup>‡</sup> More on electrical percolation in Section 3.



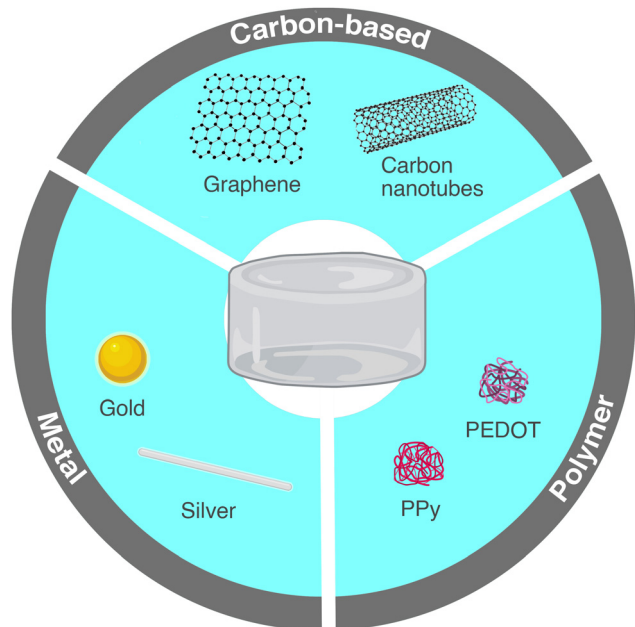


Fig. 6 Commonly used conductive nanomaterials in nanocomposite hydrogels.

carbon-based, polymeric and metals and metal oxides (Fig. 6). Throughout this review, the ISO definitions of nanomaterials and nanoparticles are used. A nanomaterial is defined as a material with any external dimension, internal structure or surface in the nanoscale, *i.e.* 1 to 100 nm. A nano-object is a discrete piece of material with one, two or three external

dimensions in the nanoscale while a nanoparticle has all three external dimensions in the nanoscale range.<sup>64</sup>

## 2.1 Carbon-based nanomaterials

Carbon-based nanomaterials have high mechanical strength, electrical conductivity, surface area and chemical stability and are available in a variety of allotropes and forms.<sup>65</sup> They can usually combine these properties, which is rather unique. The carbon nanomaterials most commonly employed to increase the electrical conductivity of hydrogels include carbon nanotubes (CNTs) and graphene-related materials (GRMs), and less often carbon black and graphite.

**2.1.1 Carbon nanotubes.** CNTs are one-dimensional, long, cylindrical nano-objects of  $sp^2$ -hybridized carbon atoms arranged in hexagonal arrays, with nanoscale diameters (Fig. 7(A)). They can be single-walled (SWNT) or multi-walled (MWNT), where multiple, concentric nanotubes with increasing diameters are held together by van der Waals forces. SWNTs can have diameters of 0.4 to 4 nm with most of them being around 1.4 nm.<sup>66</sup> Their length can range from a few hundreds of nm up to centimeters, with most nanotubes length on the micro scale.<sup>67</sup> CNTs are known to have remarkable physical properties, notably heat and electrical conductivity, mechanical strength, optical properties and a large surface area, with numerous potential applications in electronics, biomedicine, optics, composite materials and more.<sup>66–68</sup>

CNTs can be semi-conducting or metallic, depending on the diameter and the helicity of the nanotube. The electrical conductivity of isolated CNTs can reach values of  $10^5$ – $10^8$  S  $m^{-1}$ .<sup>69,70</sup>

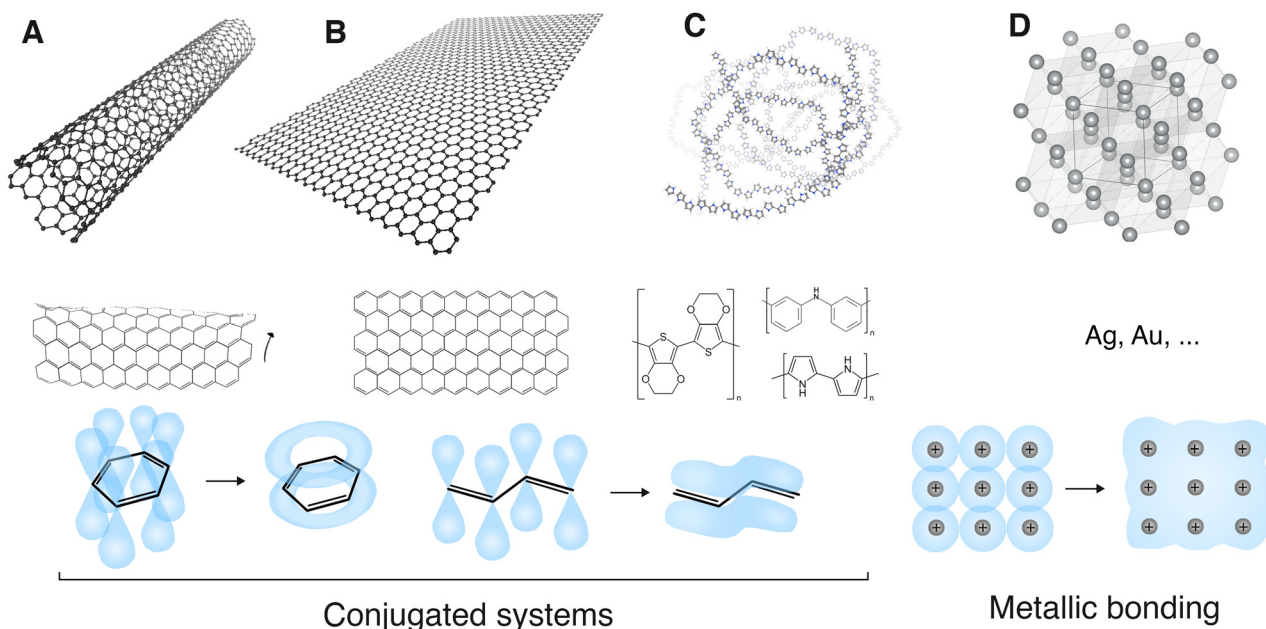


Fig. 7 Conductive nanomaterials. (A) Carbon nanotube; (B) graphene; (C) conducting polymer (polypyrrole); (D) metal (silver). A, B and C are conjugated systems. The pi orbitals of their hybridized  $sp^2$  bonds form a delocalized, overlapping and conducting pi electron orbital on a parallel plane over and under the sigma bonds. This is conventionally represented by alternating single and double bonds. D forms metallic bonding. Conduction electrons from the metal cations form a delocalized electron cloud all over the metal crystal structure. Software used: VMD<sup>81</sup> and VESTA.<sup>82</sup>





They are commonly implemented in nanocomposite hydrogels to increase their electrical conductivity because of their high intrinsic electrical conductivity, the ability to form percolation networks even at low concentration thanks to their high aspect ratio (typically from a few hundreds to tens of thousands) and the compliance with many polymer matrices.<sup>69</sup> Additionally, CNTs can also significantly reinforce the mechanical properties of nanocomposite hydrogels.<sup>71,72</sup>

Zhou *et al.* added SWNT into a gelatin hydrogel and cross-linked them with glutaraldehyde. The hydrogel served as a scaffold for engineered cardiac tissue, aimed to treat myocardial infarction. Adding 0.15% w/w of SWNT into a hydrogel with 7.5% w/w gelatin and 2.5% w/w glutaraldehyde increased the electrical conductivity from  $3 \times 10^{-8}$  to  $5 \times 10^{-5} \text{ S m}^{-1}$  and the shear modulus from 30 to 40 Pa. The SWNT concentration was chosen as a compromise between conductivity and cytotoxicity; hydrogel scaffolds with higher SWNT concentration significantly decreased cardiac cell viability. The conductive SWNTs enhanced the contractile muscle tissue function and the formation of gap junction and globally improved heart function after myocardial infarction, as tested on rats.<sup>73</sup>

Spizzirri *et al.* added MWNTs into gelatin microgels by emulsion polymerization, in the presence of sodium methacrylate and *N,N'*-ethylenebisacrylamide. The microgels served as a drug reservoir for electro-stimulated release on the skin surface. It was found that 0.8% w/w MWNTs increased the hydrogel conductivity from  $1.3 \times 10^{-7}$  to  $2.6 \times 10^{-7} \text{ S m}^{-1}$ . Lower concentrations of MWNT had no effect on the conductivity, indicating that a percolation network was not formed. The microgels with MWNTs showed no significant cytotoxic effect and increased the release rate of the drug, both with and without an external electric field.<sup>74</sup>

**2.1.2 Graphene-related materials.** Graphene is a two-dimensional, flat, one-layer sheet of  $\text{sp}^2$ -hybridized carbon atoms arranged in a hexagonal lattice (Fig. 7(B)). The production of pure, flawless graphene is complex and costly. Mechanical exfoliation of graphite produces high-quality graphene sheets, with a very low yield. Graphene has high mechanical strength and electrical conductivity. The conductivity is affected by the interactions of graphene sheets with its substrate; suspended graphene sheets have a conductivity of up to  $6 \times 10^5 \text{ S m}^{-1}$ .<sup>75</sup> Higher yields may be obtained at the cost of a lower selectivity in terms of number of layers (shifting to thicker nano-objects), ranging from few-layer graphene to multi-layer graphene and finally graphene platelets.<sup>76</sup>

Alternatively, the Hummers' method is used to oxidise graphite, which can then be exfoliated into graphene oxide (GO) in solution.<sup>77</sup> The covalent functionalization of graphene drastically reduces its conductivity to values around  $2 \times 10^{-2} \text{ S m}^{-1}$ , but GO has an easier production and can be dispersed in aqueous solutions. Then, GO can be chemically or thermally reduced to restore part of its electrical conductivity. Chemical reduction with reducing agents reaches conductivity values of  $6 \times 10^4 \text{ S m}^{-1}$  and thermal reduction with high annealing temperatures can produce reduced graphene oxide (rGO) with conductivities up to  $2 \times 10^5 \text{ S m}^{-1}$ .<sup>78</sup>

GRMs are incorporated into hydrogels to increase their electrical conductivity because of their high intrinsic conductivity and large aspect ratio, allowing for low percolation thresholds. They also simultaneously improve the mechanical properties of nanocomposite hydrogels.<sup>53</sup>

Wang *et al.* added rGO to increase the conductivity and mechanical resistance of a PVA-PDA (polyvinyl alcohol-polydopamine hydrochloride) hydrogel. The hydrogel was developed as a strain-sensitive human motion sensor. Blending in 5% GO/PVA slightly increased the conductivity from  $10^{-2}$  to  $2 \times 10^{-2} \text{ S m}^{-1}$ . However, reducing GO for 1 hour by mixing it with PDA and then adding the solution to the PVA hydrogel, in concentrations 5% w/w GO/PVA and 7.5% w/w PDA/PVA increased the conductivity of the hydrogel to  $0.1 \text{ S m}^{-1}$ . Increasing the reduction time to 4 hours further increased the conductivity to  $0.27 \text{ S m}^{-1}$ . The GO also increased the tensile strength of the hydrogels though the rGO reinforced the strength less than GO. This was attributed to the functional oxygen groups in GO, which physically interact with the polymer chains. The conductive hydrogels successfully changed their conductivity according to strain, as tested on a human wrist and knee joint.<sup>79</sup>

Alam *et al.* fabricated conductive, robust and pH-sensitive hydrogels by adding graphene to polyacrylic acid. Graphene was prepared from graphite powder, by oxidizing in an acidic environment with simultaneous ultrasonication. A thin film of the as-prepared graphene had a conductivity of  $495 \text{ S m}^{-1}$ . The composite hydrogels were prepared by *in situ* polymerization. Adding 1% v/v graphene to the hydrogels increased their electrical conductivity from  $3 \times 10^{-14}$  to  $1.3 \times 10^{-5} \text{ S m}^{-1}$ , the compressive strength from 0.4 to 6.9 MPa and the Young's modulus from 1.6 to 19 MPa.<sup>80</sup>

## 2.2 Conducting polymers

Conducting (or conjugated) polymers are organic macromolecules with a backbone of alternating double and single bonds. The delocalized pi electrons of the double bonds, in combination with a dopant ion that is added to carry charges, confer them conducting or semiconducting properties (Fig. 7(C)).<sup>83</sup> The most common conducting polymers are polypyrrole (PPy), polyaniline (PANI), polythiophene and their derivatives, notably aniline oligomers and poly(3,4-ethylenedioxythiophene) (PEDOT).<sup>22</sup>

Nanomaterials of conducting polymers can be prepared by post-polymerization dispersion or direct polymerization in disperse heterophase systems.<sup>84</sup> PPy nanomaterials are spherical with diameters ranging from 50 to 400 nm, depending on the stabilizer used and have a conductivity of  $10^2$ – $5 \times 10^3 \text{ S m}^{-1}$ .<sup>84,85</sup> PANI nanomaterials can have diameters as small as 4 nm and up to 500 nm, in spherical or ellipsoid, rice-grain shape and have a conductivity of  $10$ – $10^7 \text{ S m}^{-1}$ . PEDOT nanomaterials are often doped with anionic polyelectrolytes, such as poly(sodium-4-styrenesulfonate) (PSS), which serve as a charge balance. PEDOT:PSS nanomaterials are water soluble and have spherical size with diameters 35–100 nm and electrical conductivities of approx.  $10^5 \text{ S m}^{-1}$ .<sup>84,85</sup>



Most conducting polymers can also be directly prepared by electropolymerisation on conducting substrates.

Conducting polymers are added to hydrogels to increase their conductivity, as an alternative to carbon-based or metal nanomaterials. They are mostly inexpensive, easy to process and biocompatible.<sup>22</sup> Additionally, surface functionalization and incorporation of dopant ions can further modulate their electrical properties.<sup>83</sup> However, conducting polymers are brittle and may decrease the overall mechanical strength of nanocomposite hydrogels.<sup>86</sup>

Li *et al.* synthesized a tough and conductive nanocomposite hydrogel by *in situ* polymerization of *N*-hydroxyethyl acrylamide in a solution containing PEDOT:PSS nanoparticles and LAPO-NITE<sup>®</sup> nanoplatelets. The polymer nanoparticles were added to increase the conductivity and the clay nanoplatelets to add mechanical strength. Adding 0.5% w/v PEDOT:PSS to the hydrogel increased the conductivity from 0.04 to 0.09 S m<sup>-1</sup>. The nanocomposite hydrogels exhibited strain-responsive conductivity and self-healing capacity, making them ideal candidates for applications in motion sensors.<sup>87</sup>

Wang *et al.* prepared a gelatin methacrylate hydrogel and integrated PPy nanoparticles into it with a dopamine cross-linker. The electrical conductivity of the hydrogels increased from  $3.6 \times 10^{-3}$  S m<sup>-1</sup> (with 1 mg ml<sup>-1</sup> PPy) to  $1.2 \times 10^{-2}$  S m<sup>-1</sup> with 4 mg ml<sup>-1</sup> PPy nanoparticles. The conductive hydrogel was then tested as an engineered cardiac patch to repair myocardial infarction in affected rat models, enhancing the functionalization of cardiomyocytes.<sup>88</sup>

### 2.3 Metal and metal oxide nanomaterials

These nanomaterials are nanoscale entities of pure metals or metal compounds, such as oxides (Fig. 7(D)). The most used metal nanomaterials for increasing the electrical conductivity of hydrogels are gold and silver nano-objects. Gold nano-objects are mostly spherical or rod-shaped, have diameters of 1–60 nm and a conductivity of  $4.5 \times 10^7$  S m<sup>-1</sup>. Silver nano-objects can be spherical with diameters of 4–120 nm or nanowires, with diameters of 10–200 nm and lengths of 5–100  $\mu$ m.<sup>85,89</sup> Silver has an electrical conductivity of  $6.3 \times 10^7$  S m<sup>-1</sup>. Additionally, silver nano-objects are known to have antimicrobial properties, which can prove useful for some applications.<sup>90</sup> One major drawback of metal nano-objects is their propensity to oxidation, leading to a decrease in performance over time.

Baei *et al.* synthesized a thermosensitive, conductive hydrogel by embedding gold nanoparticles into a chitosan matrix. The gold nanoparticles were spherical with an average diameter of 7 nm. A hydrogel with 0.016% w/v gold nanoparticles had a conductivity of 0.13 S m<sup>-1</sup>, close the conductivity of the native myocardium (0.16 S m<sup>-1</sup>), while the pristine chitosan hydrogel was not conductive. The conductive nanocomposite hydrogel stimulated the differentiation of mesenchymal stem cells into cardiomyocytes and was deemed promising for use as an injectable hydrogel to deliver cells and bioactive factors to the infarcted heart.<sup>91</sup>

### 2.4 MXenes

MXenes, two-dimensional sheets of transition metal carbides, carbonitrides or nitrides, represent a more recent class of conductive nanomaterials.<sup>92</sup> MXenes are prepared by selective chemical etching of the A element from MAX phases, layered ternary solids with a general formula of M<sub>n+1</sub>AX<sub>n</sub> ( $n = 1-4$ ) where M is an early d-block transition metal, A is a main-group sp element and X is C, N or both.<sup>93</sup> They consist of sheets with a thickness of few atomic layers (3 to 9 atomic layers) and up to tens of microns in their lateral dimensions.<sup>92,94</sup> MXenes can have semiconducting or metallic conductivities with the highest values reported being  $2.4 \times 10^6$  S m<sup>-1</sup> for a Ti<sub>3</sub>C<sub>2</sub> MXene.<sup>95</sup> MXene surfaces are commonly terminated with oxygen or hydroxyl groups, which render them hydrophilic.<sup>92,94</sup> Their hydrophilicity, high electrical conductivity and large aspect ratio make them suitable candidates for incorporation in conductive nanocomposite hydrogels.<sup>94</sup> For the moment, notable biomedical applications of MXene nanocomposite hydrogels have mostly focused on strain sensors.<sup>94,96</sup> An important limitation is that MXenes have a metastable character in aqueous media, *i.e.* they are prone to degradation.<sup>96,97</sup>

## 3. Electrical percolation

### 3.1 Percolation theory

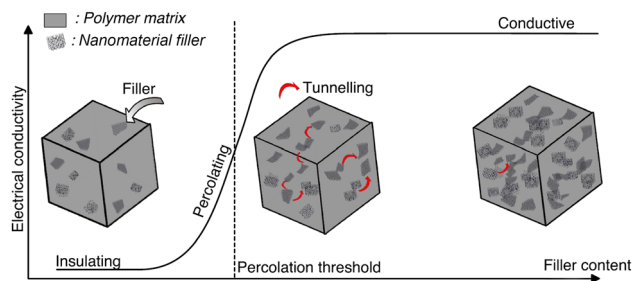
Incorporation of a conducting filler in an insulating hydrogel matrix, increases the conductivity of the nanocomposite, with increasing filler concentration. Often, a point is observed where a small increase in filler concentration leads to a steep increase in conductivity, a shift of few orders of magnitude.<sup>69,98</sup> Further increase in filler concentration has a limited effect on conductivity. The resultant graph of conductivity *versus* filler concentration exhibits a sigmoidal shape (Fig. 8).

Percolation theory explains this jump-like transition. The conductivity increases steeply when the filler forms a continuous conducting network within the hydrogel. Individual clusters of filler particles come into contact with each other, to ultimately form a single cluster that extends throughout the system.<sup>98</sup> Electron transport takes place through the conductive filler network and the system's behaviour changes from insulating to conducting (Fig. 8). The corresponding critical filler concentration is called percolation threshold. The percolation threshold of different nanocomposite hydrogels varies widely and depends on the polymer matrix and the shape, size, orientation and dispersion of the filler.<sup>69</sup> Higher aspect ratios and better dispersions lead to lower percolation thresholds.

The electrical percolation threshold is an important parameter in percolating systems. Experimentally finding it and tuning it by changing the properties of the system can lead to the development of highly conductive nanocomposites with minimal filler concentration and desirable properties. For a random distribution of a well-dispersed filler, the statistical percolation theory can model the conductivity of composites.<sup>98,99</sup>

$$\sigma = \sigma_0(\Phi - \Phi_c)^t, \text{ for } \Phi > \Phi_c \quad (1)$$





**Fig. 8** Electrical conductivity vs. nanofiller content. Percolation has three phases: initially the concentration of the nanomaterial is not enough to form an interconnecting network and the conductivity remains low. As the concentration increases, electrons can hop through the nanofillers with tunnelling and the conductivity increases. When the concentration reaches the percolation threshold, a network is formed throughout the whole system and the conductivity reaches a high plateau. Adapted from ref. 78.

where  $\sigma$  is the electrical conductivity of the composite,  $\sigma_0$  the electrical conductivity of the filler in its bulk form,  $\Phi$  the filler volume fraction,  $\Phi_c$  the percolation threshold and  $t$  the critical power law exponent. The exponent  $t$  depends on the system dimensionality and takes values of *ca.* 1.3 for 2D systems and *ca.* 2 for 3D.<sup>78,100</sup> The electrical percolation threshold can be determined experimentally or numerically.

Experimentally, the electrical percolation threshold can be deduced from a graph of electrical conductivity with increasing filler concentration. The threshold is the middle point of the S-shaped part of the graph. It can also be calculated by fitting eqn (1) to experimental data.

Numerically, there are various models with increasing complexity for calculating the percolation threshold. The first studies of percolation assumed a lattice organisation of particles where a set of predefined spatial positions may be covered or be left empty.<sup>98</sup> In materials science though, the results of these studies can only be relevant for systems with a crystalline structure. The continuum percolation models can be applied for hydrogels, which have an amorphous structure and thus a random distribution of particles. The simplest models simulate the filler particles with interpenetrating objects. The electrical percolation threshold is calculated as the point where the filler particles form a continuous network, a cluster of particles that extends through the simulation space.

The calculated percolation thresholds from the above method can be higher than the experimental ones because it does not take into account two phenomena: the filler particles cannot penetrate into each other because of repulsive van der Waals interactions and the electrical percolation threshold can occur before the geometrical percolation, due to electron tunnelling.<sup>101</sup> Geometrical percolation is when the filler particles form a network with physical contact. Electrical percolation can occur at lower filler concentrations, as electrons can be transferred through a thin film of dielectric material that separates the filler particles. The electron tunnelling distance is in the order of few nm.<sup>102</sup>

The hard-core, soft-shell simulation model is used to represent more accurately the nanocomposite system. In this

model, the fillers are randomly distributed in a fixed space and are represented by impenetrable hard cores. The soft shells can overlap with the cores and with each other and represent the tunnelling distance.<sup>69,101</sup> Another factor that differentiates experimental from numerical results is the non-randomness of real dispersions. In numerical models, the particles will be randomly distributed within the system's boundaries while in experiments particles tend to agglomerate/cluster and align, affected by particle interactions, the dispersion method and the thermal or mechanical processing history of the material.<sup>103</sup> This clustering and non-random alignment of nanomaterials can have substantial effects on the percolation concentration of the system.<sup>103,104</sup>

### 3.2 Critical path approximation

Percolation models assume a sharp cut-off point of electrical conductivity. Two individual particles are either electrically connected or not. Subsequently, a cluster of particles passes from disconnected (insulating) to percolated (electrically conductive), when the percolation threshold is reached. Ambrosetti *et al.* argue that this approach is well suited to explain the electrical conductivity in the extreme cases of low filler concentration (particles with no electrical contact) and high filler concentration (particles “touching” each other throughout the system) but fails to account for the conductivity changes in the intermediate regime, around the percolation concentration.<sup>105</sup> They model the conductivity changes in nanocomposite systems by focusing on the tunnelling conductance between conductive particles.<sup>105,106</sup> The tunnelling conductance decays exponentially with distance, but does not include a sharp cut-off. This model can be solved numerically by simulating the conducting fillers as a network of particles that are all connected to each other through tunnelling processes (Global Tunnelling Network). An analytical solution for the conductivity  $\sigma$ , is given by the critical path approximation

$$\sigma \cong \sigma_0 \exp \left[ -\frac{2\delta_c(\varphi, a, b)}{\xi} \right] \quad (2)$$

where  $\sigma_0$  a constant,  $\xi$  the characteristic tunnelling length and  $\delta_c$  a critical distance, which depends on the filler concentration  $\varphi$  and the geometric characteristics of the particles  $a$  and  $b$  ( $a/b$  is the aspect ratio). The solution of eqn (2) reduces the conductivity of a nanocomposite system to the calculation of the geometrical parameter  $\delta_c$  and is in good accordance with the numerical solutions from the global tunnelling network model.<sup>105</sup>

The implications of the critical path approximation are that the transition from insulating to conducting is no longer described by a power law increase in conductivity after the percolation threshold concentration, but rather as a crossover between the insulating matrix conductivity and the interparticle tunnelling conductivity.<sup>105</sup>

### 3.3 Percolation of carbon nanotubes

CNTs have a high aspect ratio (generally *ca.* 1000 or higher), which allows for low percolation thresholds.<sup>69</sup> Kovacs *et al.* used the excluded volume concept to calculate a percolation





threshold of

$$\Phi_c = \frac{1}{\eta} = \frac{1}{1000} = 0.1\% \text{w/w} \quad (3)$$

where  $\eta$  is the aspect ratio of CNTs.<sup>100</sup> They argue that this percolation threshold is universal for CNTs in insulating polymer matrices (they reviewed solid nanocomposite polymers but their results are useful in the case of hydrogels too). Deviations with higher  $\Phi_c$  are attributed to poor dispersion and lower  $\Phi_c$  are attributed to kinetic percolation, a state where the particles are free to move through diffusion, convection, shearing or external fields and form a conducting network at lower concentrations. The critical exponent  $t$  for CNTs, calculated from fitting the experimental data into eqn (1) ranged from 0.9 to 7.6, peaking at  $t = 2$ .<sup>100</sup>

CNTs can be modelled as capped cylinders for the numerical simulations. The most sophisticated models take into account the electron tunneling distance, the non-random alignment, as well as the waviness of the nanotubes, which increases the percolation threshold.<sup>107</sup>

### 3.4 Percolation of graphene-related materials

The high specific surface area and aspect ratio of GRMs allow for low percolation thresholds. Zhang *et al.* compared the percolation threshold of graphene and graphite nanocomposites, to illustrate the influence of the aspect ratio and specific surface. They measured the conductivity of PET/graphene and PET/graphite with increasing filler concentrations and obtained a percolation threshold of 0.47% v/v for graphene and 3.5% v/v for graphite.<sup>108</sup>

Marsden *et al.* reviewed the electrical percolation threshold of GRM nanocomposites and the average percolation threshold was 0.5% v/v.<sup>78</sup> A high standard deviation though (0.7% v/v) indicates that the percolation threshold may vary widely according to the dispersion, shape parameters of the filler (aspect ratio and specific surface) and the polymer matrix.

### 3.5 Electrical percolation studies in hydrogels

Most conductive hydrogel studies presented in this review did not report a percolation threshold. The ones commenting on percolation threshold concentration are grouped here. Ferris and Panhuis found a percolation threshold of 1.3% w/w for gellan gum hydrogels containing multi-walled CNTs.<sup>109</sup> Mottet *et al.* report a percolation threshold of 0.5% w/w for alginate hydrogels with CNTs.<sup>110</sup> Cui *et al.* report a percolation threshold of 0.015% w/w for poly-ethyl acrylate hydrogels, with multi-walled CNTs.<sup>111</sup> Guillet *et al.* and Macdonald *et al.* found no percolation threshold for agarose – double-wall CNTs hydrogels for concentrations up to 1% w/w and for collagen – SWNTs for concentrations up to 0.008% w/w respectively.<sup>112,113</sup> Alam *et al.* reported a percolation threshold of 0.4% v/v for graphene incorporated in poly acrylic acid hydrogels.<sup>80</sup> Sayyar *et al.* and Qiu *et al.* both reported a percolation threshold of 0.1% w/w for rGO fillers in chitosan and poly-isopropyl acrylamide hydrogels.<sup>114,115</sup>

In contrast, in studies of solid nanocomposite polymers, there are more publications reporting a clear electrical percolation threshold. We explain this by four factors: (1) there are fewer studies in total for nanocomposite hydrogels, (2) carbon-based materials which generally exhibit low percolation thresholds are highly hydrophobic resulting in the formation of aggregates and poor dispersions in aqueous media, (3) polymer chains may tend to wrap around nano-objects, limiting direct contact between the conductive phase, and (4) hydrogels are more complex systems consisting of at least three components (polymer matrix, water, filler) and two conduction mechanisms (ionic and electronic). This complicates the investigation of the percolation threshold. Rather than a clear cut-off point of several orders of magnitude increase in conductivity, many nanocomposite hydrogels present modest augmentations with increasing nanofiller content.

### 3.6 Dispersion

Nanomaterials are introduced into nanocomposite hydrogels using one of three possible approaches: (1) dispersion of nanomaterials in an aqueous suspension of a monomer, followed by gelation, (2) dispersion of a nanomaterial precursor in an aqueous suspension of a monomer, followed by gelation and nanomaterial synthesis within the polymerized matrix, and (3) physical embedding of nanomaterials into a hydrogel matrix.<sup>116,117</sup> The nanomaterials can significantly alter the properties of the nanocomposite hydrogels, thanks to the multiple physical and/or chemical interactions between the nanomaterials and the polymer. These include hydrogen bonds, van der Waals interactions and electrostatic interactions.<sup>116</sup> The dispersion of nanomaterials in the hydrogel network affects the system's electrical and mechanical properties. Poor dispersions, leading to nanomaterial agglomeration undermine the property-enhancing features of nano-engineering.

Carbon-based nanomaterials have highly hydrophobic surfaces. They form agglomerates in aqueous suspensions and have low interfacial compatibility with polymer matrices.<sup>116,118</sup> The surface of carbon nanotubes and graphene sheets can be functionalized with hydrophilic groups, such as –COOH and –OH, facilitating their dispersion in water through electrostatic repulsion (at slightly acidic pH and above, the carboxylic function is present as a negatively charged carboxylate).<sup>119</sup> However, covalent functionalization disrupts the electronic structure of pristine carbon-based nanomaterials, deteriorating their electrical conductivity.<sup>116</sup> Another route is the addition of surfactants. Lastly, physical methods, such as polymer wrapping and cellulose-assisted dispersion have also been developed. Polymer wrapping and surfactants improve dispersion but also cover the surface of carbon nanomaterials and may increase the distance between the nanomaterials, decreasing electron tunnelling conductance and negatively affecting the conductivity of the nanocomposite hydrogel.<sup>118</sup>

Metal nanomaterials have high surface energy and strong dipole–dipole attractions. They form aggregates and precipitate in aqueous suspensions. Two approaches that achieve stable dispersions with fewer aggregates are the *in situ* growth of metal nanomaterials after gelation of the hydrogel and the use



of metal nanomaterials as reactive cross-linking agents to polymerize the hydrogel.<sup>116</sup>

Conducting polymers are hydrophobic in their undoped form.<sup>83</sup> Nanoparticles of conducting polymer, stabilized with a dopant, can be dispersed in aqueous solutions by stirring and sonicating. PEDOT is a hydrophobic, positively charged conjugated polymer. It is usually doped with PSS, a hydrophilic, negatively charged polyelectrolyte, to create stable water dispersions of PEDOT:PSS.<sup>120</sup> PEDOT:PSS nanoparticles have fairly good dispersibility in water.<sup>87</sup> Similarly, PANI is hydrophobic in its undoped form and hydrophilic when stabilized with a dopant such as camphor sulfonic acid.<sup>121,122</sup> Polypyrrole nanoparticles can be doped with iron(III) chloride or PSS and can be easily dispersed in aqueous solutions.<sup>123,124</sup>

## 4. Electrical conductivity and conduction mechanisms

### 4.1 Conductivity

Electrical conductivity is an intrinsic property of materials which measures how well they conduct electric current. It is the reciprocal of resistivity, the resistance of a material to the flow of electric current. When a potential difference is applied to a sample, an electric field  $E$  is created, and charges inside the sample have an electric force applied to them. For an isotropic conductivity and a homogenous electric field, the current density  $J$  that results, depends on the sample's conductivity  $\sigma$ .

$$J = \sigma \cdot E \quad (4)$$

Electric current is conducted by charge carriers, which can be ions, electrons and holes. Electrical conductivity depends on the product of charge carrier concentration and mobility.

### 4.2 Conduction mechanisms

When a sample is placed between two electrodes, charge is injected in it through the interface of the sample with the electrode materials, then passes through its volume and gets out again from the interface of the sample with the other electrode. Localised charge movements also result in a measurable current, even if the charge carrier does not exit the sample. The limiting conduction mechanism is the determining one and this can fall into one of two categories: interface-limited (or electrode-limited) and volume-limited (or bulk-limited).<sup>125</sup> Depending on the nature of the material and the applied electric field, one or more conduction mechanisms may contribute to the electric current transport.

Interface-limited conduction mechanisms depend on the electrical properties at the contact between the sample and the electrode. The most important parameters are the energy barrier height that the charge carrier has to overcome in order to get injected into the sample and the effective mass of the charge carriers.<sup>125,126</sup> Volume-limited conduction mechanisms depend on the electrical properties of the sample. The most important parameter in volume-limited conduction are traps, locations in solids which restrict the movement of charge

Table 1 Conduction mechanisms in solids and equations<sup>112,127</sup>

Volume-limited		Interface-limited	
Ohmic	$J = \sigma E$	Schottky	$J = AT^2 e^{\left(\frac{\phi_0 - \beta_{\text{sv}} \sqrt{E}}{k_B T}\right)}$
Ionic (Hopping)	$J = J_0 \sinh\left(\frac{q l_i E}{k_B T}\right)$	Fowler–Nordheim–Tunneling effect	$J = AE^2 e^{\left(-\frac{\beta}{E}\right)}$
Poole–Frenkel	$J = J_0 e^{\frac{\phi_0 - \beta_{\text{PF}} \sqrt{E}}{k_B T}}$		

carriers. Trap energy level, trap spacing and concentration all affect volume-limited conduction. Other parameters include the electrical mobility, the dielectric relaxation time and the density of states in the conduction band.<sup>125,126</sup> Table 1 summarizes the conduction mechanisms in solid materials and their equations, where  $J$  the current density,  $\sigma$  the conductivity,  $E$  the electric field,  $J_0$  the zero-field current density,  $q$  the electric charge,  $l_i$  the ion jump distance from one potential well to another,  $k_B$  the Boltzmann constant,  $T$  the temperature,  $\phi_0$  the barrier height and  $A$  and  $\beta$  constants.

### 4.3 Ionic conduction

Ionic conduction occurs when the charge carriers are ions. It is the dominant conduction mechanism in electrolyte solutions but also contributes to the electrical conductivity of solids. Fast ion conductors and solid electrolytes are examples of solid materials where ionic conduction is the main conduction mechanism.<sup>128</sup> The ionic conduction mechanism consists of a series of jumps over potential barriers from one site to another.<sup>126</sup> Ions are several orders of magnitude bigger and heavier than electrons and consequently ion mobility, reduced by steric effects, is much lower than electron mobility. Ionic conduction depends on ion concentration, temperature, ion size and valency, electrical field magnitude, viscosity (in the case of solutions) and the height and spacing of potential barriers (in the case of solids).

### 4.4 Electric current

The movement of charge carriers under the influence of an electric field is the conduction current. There are two more current types that can be measured. The diffusion current is created by the movement of charge carriers under the influence of their concentration gradient. The displacement current is a transitory current due to the variation of the electric field and is not linked to a charge movement inside the sample. It includes the polarisation current, a transitory current that arises from the orientation of dipoles with the electric field. The total current can be written as the sum of the above current types

$$j(t) = \underbrace{qn(x,t) \mu(E,t) E(x,t)}_{\text{Conduction}} + \underbrace{-qD_n \frac{\partial n(x,t)}{\partial tx}}_{\text{Diffusion}} + \underbrace{\epsilon_0 \frac{\partial E}{\partial t} + \frac{\partial P}{\partial t}}_{\text{Displacement}} + \underbrace{\epsilon_0 \frac{\partial E}{\partial t}}_{\text{Polarisation}} \quad (5)$$

where  $j$  is the current density,  $q$  the elemental charge,  $n$  the charge carrier concentration,  $\mu$  the charge carrier mobility,  $E$



the electric field  $D_n$  the diffusion constant of the charge carriers,  $\epsilon_0$  the permittivity of vacuum and  $P$  is polarisation.

#### 4.5 Hydrogel water content and electrical conductivity

Nanocomposite hydrogels are complex systems consisting of at least three components: water, which comprises the largest part of the hydrogel, a cross-linked polymer network and the filler nanomaterials. The role of each constituent in the electrical conductivity of the system is still poorly understood. Deionised water is used for the fabrication of most hydrogel samples. Nevertheless, some ions will always be present, introduced to the system from the polymer, the nanomaterials and other impurities, the contact with air and labware and due to the non-perfect deionisation of water. These free ions contribute to the electrical conductivity of nanocomposite hydrogels by ionic conduction. However, the mobility of ions in hydrogels is restricted by the porous network of the polymer. The ion mobility inside a hydrogel depends on the concentration and charge of the polymer and the pore size and distribution and is significantly lower than the ion mobility in an aqueous solution. The total amount of water within hydrogels can be classified into three types, according to phase transition behaviour: (1) non-freezing (bound) water, in the primary hydration shell of the hydrophilic polymer chains, (2) freezing bound water, in the secondary hydration shell and (3) freezing free water, which does not interact with the polymer matrix.<sup>129,130</sup> These different states of water within the hydrogel framework can be distinguished through differential scanning calorimetry (DSC).<sup>131</sup> Ionic mobility is expected to be much lower in the bound water fraction.<sup>132,133</sup> Therefore, the ionic conductivity is mostly affected by the free water in the hydrogel matrix. The polymer network consists of electrically insulating polymer chains. The conduction mechanisms through it are expected to be equivalent to conduction through non-crystalline dielectric solids. Lastly, the nanomaterials studied in this review are embedded in hydrogels because of their metallic conduction properties. When they are present in a critically high concentration, they form a percolating network through the hydrogel allowing for a steep increase in the system's conductivity. The conduction mechanism through a percolating network of conductive nanomaterials is ohmic conduction.<sup>112</sup>

The water content of nanocomposite hydrogels directly affects their electrical conductivity. Lower water contents decrease ionic conduction by densifying the polymer network, leading to smaller pores and lower ion mobility. At the same time, a decrease in water content with the subsequent densification of the hydrogel, increases the volume fraction of the nanomaterials and thus the chances of forming a percolating network. Ferris and Panhuis measured the electrical conductivity of a gellan gum hydrogel embedded with multi-walled CNTs and a control gellan gum hydrogel (without nanomaterials), upon drying. They inserted the hydrogels between two electrodes, put it on a mass balance and placed all the system in a heated chamber, with the temperature increasing from 20 to 60 °C, over time. The device allowed for simultaneous mass and conductivity measurements. The initial conductivity for both

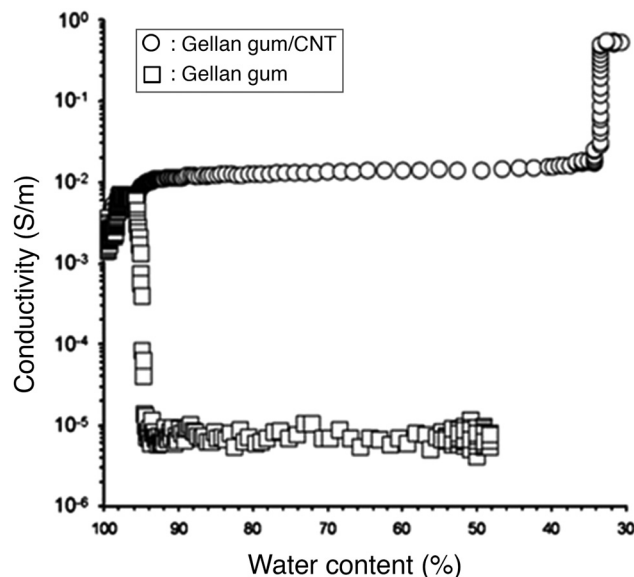


Fig. 9 Conductivity of gellan gum hydrogels with CNTs (circles) and without (squares), over decreasing water content. The loss of water content densifies the polymer network restricting ion mobility but simultaneously increases the nanomaterial volume fraction and leads to percolation. Adapted from ref. 109 with permission from The Royal Society of Chemistry.

hydrogels was  $10^{-3} \text{ S m}^{-1}$ . At 95% water content the conductivity of the nanocomposite hydrogel increased to  $10^{-2} \text{ S m}^{-1}$  while the conductivity of the control hydrogel decreased dramatically to  $10^{-5} \text{ S m}^{-1}$  (Fig. 9).<sup>109</sup> The initial hydrogel conductivity can be attributed to ionic conduction, while the differences in the behaviours for decreasing water content can be explained on the basis of electrical percolation of the CNTs and ion mobility restriction in the control hydrogel.

## 5. Measuring electrical properties

The electrical conductivity of hydrogels can be measured from the relation between voltage and current. First, the sample hydrogel is placed between two electrodes. Then, an electrical stimulus is applied to the electrodes (known voltage or current) and the response of the system is observed. The electrical response of the system depends on the transport of charges within the material and the transfer of electrons to or from the atoms and ions of the sample at the electrode-sample interface. The total current flow will depend on the resistance of the sample and the reaction rates at the interface between the sample and electrodes. In addition, the electrical response will be affected by impurities on the sample and electrodes, possibly leading to oxidation or reduction reactions. Typically, these measures apply low voltages to the sample (50 mV to 1 V). Care has to be taken to not apply over 1.23 V for an extended time period, which would lead to electrolysis of water molecules (oxidation, release of oxygen) and affect the results.

### 5.1 Two-point and four-point probe methods

The measuring device can be connected to the electrodes *via* a two-point or four-point system. Two-point probes are easier to





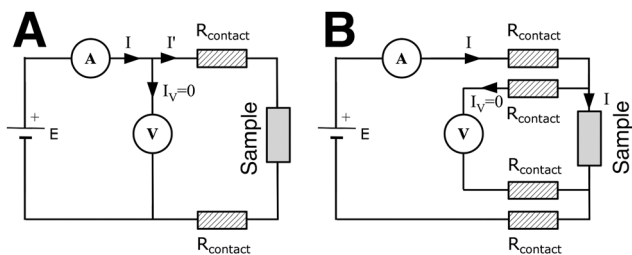


Fig. 10 Circuit diagrams of (A) two-point probe and (B) four-point probe method.

handle but also measure the contact resistance, *i.e.* the resistance at the interface between the cables and the electrodes (Fig. 10(A)). A four-point system measures directly the resistance of the sample. The electrical stimulus is applied through two probes and the sample's response is measured through the two other probes, avoiding the contact resistance (Fig. 10(B)). However, in most cases of nanocomposite hydrogels, the sample's resistance is several orders of magnitude higher than the contact resistance, so the choice of a two- or four-point probe is of negligible impact.

## 5.2 Direct current

**5.2.1 Ohmmeter.** An ohmmeter calculates the resistance  $R$  of a sample by applying a fixed DC current of few mA and measuring the corresponding voltage.

$$R = \frac{V}{I} \quad (6)$$

where  $V$  is the measured tension and  $I$  the applied current.

For an isotropic conductivity and a homogenous electric field, and with a simple sample geometry, the conductivity is calculated as the reciprocal of resistivity.

$$\left. \begin{aligned} R &= \rho \frac{l}{A} \\ \sigma &= \frac{1}{\rho} \end{aligned} \right\} \Rightarrow \sigma = \frac{l}{A} \cdot \frac{1}{R} \quad (7)$$

where  $\rho$  the resistivity,  $l$  the length and  $A$  the surface of the sample. DC conductivity measurement with a multimeter is the most simple and straightforward way to measure the conductivity of a sample hydrogel. However, it doesn't provide any information about capacitive and inductive behaviour. Most works reviewed here use this measurement, since they only seek to demonstrate an increased electrical conductivity through the incorporation of nanomaterials.

**5.2.2 Chronoamperometry.** In chronoamperometry, the electrical stimulus applied to the electrodes is a square-wave potential. The electrical response of the system is the current as a function of time  $I(t)$ . By applying a step potential difference on a hydrogel sample, a peak of current will be measured. This peak corresponds to the sum of the conduction current and the polarisation current, *i.e.* the orientation of water and other polar molecules in the direction of the electric field. The polarisation current exponentially decays and the current then reaches a plateau corresponding to the conduction current, *i.e.*

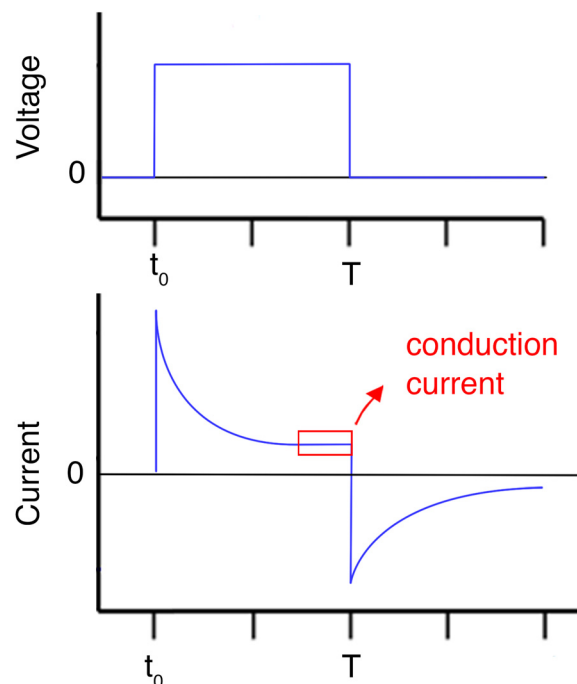


Fig. 11 Double-pulsed, controlled potential chronoamperometry. A square wave voltage initially causes a current peak, due to polarisation. The current then exponentially decays to the conduction current. The removal of the voltage may cause a brief reversed current peak.

the movement of ions and electrons within the hydrogel. When the step voltage is removed, the measured current may briefly exhibit a reverse peak, before decaying to zero. This reverse current is the result of the return of the electric dipoles (polar molecules) to a random orientation, after the removal of the electric field (Fig. 11).

From eqn (4), we have  $\sigma = \frac{J}{E}$  and for a simple geometry with the normal of the electric field perpendicular to it

$$\sigma = \frac{J}{E} = \frac{I/A}{V/l} = \frac{I \cdot l}{V \cdot A} \quad (8)$$

where  $V$  is the tension,  $I$  the current,  $l$  the length and  $A$  the surface of the sample hydrogel.

To calculate the conductivity, the conduction current (the time-invariant part of the  $I(t)$  graph) is inserted into eqn (8). Chronoamperometry also gives additional info about the sample's RC time constant.

Guillet *et al.* studied the conduction mechanism of agarose/CNTs nanocomposite hydrogels through chronoamperometry. They applied increasing voltage from 50 mV to 1.3 V and measured the current passing through the sample, *vs.* time. By plotting the conduction current density *vs.* the electric field (applied voltage divided by sample thickness) and comparing the graph with known equations of conduction mechanisms (Table 1), they suggested that pristine agarose hydrogels are dominated by an ionic conduction type, while nanocomposite hydrogels with CNTs most possibly exhibit a Poole-Frenkel conduction type.<sup>112</sup>



### 5.3 Alternating current

**5.3.1 Impedance spectroscopy.** In impedance spectroscopy, the electrical stimulus applied to the sample is a sinusoidal alternating voltage, over a range of frequencies (commonly between 0.1 and  $10^6$  Hz). The measured response of the system is the resulting current and the phase difference between signal and response.<sup>134</sup> For an electrical stimulus of  $v(t) = V_m \sin(\omega t)$ , with a frequency  $f = \omega/2\pi$ , the resulting current  $i(t) = I_m \sin(\omega t + \theta)$  is measured (for a linear behaviour). Here,  $\theta$  is the phase shift between the voltage and the current. It is equal to 0 for a purely resistive behaviour,  $\pi/2$  for a purely inductive and  $-\pi/2$  for purely capacitive. The conductive nanomaterials incorporated in hydrogels have a metallic conductivity, therefore phase shifts close to 0 degrees. In pristine hydrogels, ionic conductivity dominates and the values of  $\theta$  are near  $-30$  to  $-40$  degrees.

From the amplitude and phase shift measurement, a complex impedance  $Z$  is deduced with  $Z'$  the real part (resistance) and  $Z''$  the imaginary part (reactance). Impedance expands the notion of resistance by taking reactance into account. While resistance leads to the dissipation of energy as heat, reactance stores energy and releases it after  $\pi/2$ . Capacitive reactance stores energy in the form of an electric field and inductive reactance stores energy in the form of a magnetic field.

The analysis of impedance spectroscopy data provides information about the electrical properties of the sample hydrogel. Different excitation frequencies will elicit different electrical responses from the material. For example, in low frequencies, electrons, ions and dipoles of different sizes will all move responding to the electric field. In higher frequencies ions and bigger dipoles do not have the time to move; only electrons respond to a rapidly changing electric field. Plotting the impedance data into a Nyquist plot (Fig. 12) allows the visualization of electrical phenomena and the modelling of an equivalent circuit, *i.e.* a simplified theoretical model that retains all the electrical characteristics of the original, complex circuit.<sup>135</sup> The y-axis represents the negative of the imaginary part and the x-axis represents the real part of the complex impedance. The intersection of the curve with the x-axis designates the

resistance of the hydrogel.<sup>136</sup> The conductivity is then calculated from eqn (7).

In the case of nanocomposite hydrogels, the equivalent circuit seems to include a Warburg impedance element in series with a resistor.<sup>112,137,138</sup> The Warburg element models diffusion processes and is recognizable by a straight line with  $45^\circ$  slope, at low frequencies (Fig. 12(B)).

Warren *et al.* used impedance spectroscopy to investigate the percolation behaviour of CNT-loaded gellan gum hydrogels.<sup>137</sup> They prepared the hydrogels by dissolving gellan gum powder into warm ( $80^\circ\text{C}$ ) deionised water, dispersing CNTs in the solution through sonolysis and then cross-linking with  $\text{Ca}^{2+}$  ions. By varying the length of the hydrogel, they were able to distinguish between the sample's resistance and the contact resistance: the sample's resistance increases linearly with length, while the contact resistance remains invariable.

$$R_{\text{measured}} = \frac{1}{\sigma_{\text{sample}}} \frac{l}{A} + R_{\text{contact}} \quad (9)$$

where  $\sigma_{\text{sample}}$  the conductivity of the sample (hydrogel),  $l$  the length and  $A$  the cross section.

Three different carbon nanomaterials were employed, single-wall CNTs, multi-wall CNTs and vapour-grown carbon nanofiber. The impedance analysis for all three carbon nanomaterials in a 0.9% v/v concentration showed a similar modest increase in conductivity (final value  $0.12 \text{ S m}^{-1}$ ; the original conductivity of gellan gum is not mentioned here but is reported as  $10^{-3} \text{ S m}^{-1}$  in previous studies<sup>109</sup>). This concentration was deemed too low to have an effect on the conduction mechanism. The carbon filler concentration was further increased by selectively removing a part of the water content of the hydrogel, at a rate of  $0.43 \text{ g h}^{-1}$ , in a controlled temperature and humidity chamber. It was reported that at MWNT concentration of 1.4% v/v the impedance was no longer dependent on frequency and the Warburg coefficient value rapidly decreased, witnessing the formation of a percolating network and the transition between transport dominated by ions to transport dominated by electrons.<sup>137</sup>

**5.3.2 Conductivity meter.** Another way to measure the electrical conductivity of hydrogels is the use of a conductivity meter with an adequate probe. These devices are optimized for fast conductivity measurements of aqueous, ionic solutions but can also work in the case of hydrogels. In conductivity meters, the electrical stimulus applied is a single-frequency alternating current (or voltage) and the produced voltage (or current) is read. The conductivity meters are calibrated with solutions of known conductivity and their output is directly the conductivity value. Koppes *et al.* used a conductivity meter to evaluate changes in conductivity induced by the incorporation of single-wall CNTs into collagen type I hydrogels. For the measurements, they submerged the conductivity meter probe in the hydrogel precursor solution, cross-linked the hydrogel and then measured the conductivity. A concentration of 0.01% w/w of CNT increased the conductivity of collagen hydrogels from  $1.4$  to  $2.4 \text{ S m}^{-1}$ .<sup>139</sup>

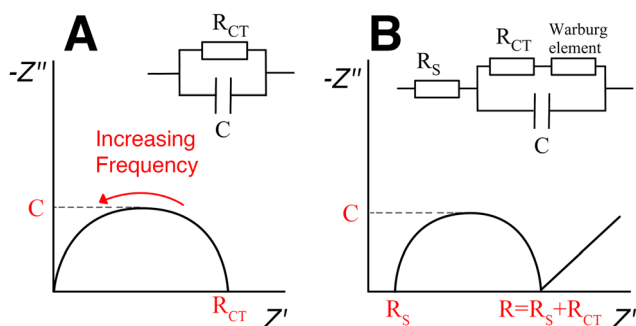


Fig. 12 Impedance spectroscopy results can be visualised with a Nyquist plot. (A) Typical Nyquist plot of simple RC circuit with one resistor and one capacitor in parallel. (B) Typical Nyquist plot of resistance in series with the parallel combination of a capacitance with a resistance and a Warburg diffusion element (also known as Randles circuit).



### 5.4 Hall effect

A Hall effect measurement system can be also used to measure a hydrogel's conductivity. When a sample is traversed by an electric current in one direction, and a magnetic field perpendicular to this direction is applied simultaneously, a potential difference will be produced along the sample.<sup>125,140</sup> This potential difference is called the Hall voltage. The Hall effect measurement also provides information about the charge carrier concentration and mobility.<sup>125</sup> Bu *et al.* prepared a sodium alginate and carboxymethyl chitosan hydrogel to use as a substrate for peripheral nerve regeneration. They doped the hydrogel with conductive PPy (In this case, not strictly a nanocomposite but a hybrid hydrogel) and measured its conductivity through a Hall effect testing system. The conductivity of the hydrogel increased from  $7.35 \times 10^{-6}$  to  $8 \times 10^{-3} \text{ S m}^{-1}$  by increasing the PPy mass ratio from 0.02 to 0.4, but no information was given on the charge carrier concentration and mobility.<sup>86</sup>

### 5.5 Conductive atomic force microscopy

Conductive atomic force microscopy (C-AFM) is a mode in AFM that allows for simultaneous measurement of the topography of a material and the electric current flow at the contact point of the probe with the sample.<sup>141</sup> In nanomaterials engineering, it can be used to evaluate the property-enhancement and dispersion of conductive nanomaterials on the surface of a sample. Annabi *et al.* fabricated an elastic, conductive hydrogel based on a human recombinant protein and GO. They dispersed  $2 \text{ mg ml}^{-1}$  GO nanomaterials in a methacryloyl-substituted tropoelastin prepolymer solution and then initiated UV photocrosslinking to obtain a biocompatible, highly elastic hydrogel for the regeneration of electroactive tissue. They conducted C-AFM measurements on a pristine and a nanocomposite hydrogel (Fig. 13). Impedance spectroscopy, C-AFM and excitation threshold measurements all

confirmed the higher electrical conductivity of the nanocomposite hydrogel.<sup>142</sup>

## 6. Discussion

### 6.1 Measured electrical conductivities of nanocomposite hydrogels

Conductive nanocomposite hydrogels are a relatively recent but rapidly expanding field of research. Virtually all publications covered in this review were published in the last 10 years. The main application that drives research are electrically conductive scaffolds for tissue engineering of electroactive cells. Other applications include strain sensors, biosensors, drug delivery, artificial skin, electrocardiography gel and electromagnetic interference shielding.<sup>74,143–145</sup> Table 2 presents a comprehensive summary of nanocomposite electrically conductive hydrogels, along with the nanomaterial concentrations employed and the highest conductivity value measured.

Impedance spectroscopy graphs give impedance or conductivity values over a range of AC frequencies. In some cases, when the low-frequency end of the graph forms a straight line, a DC conductivity value can be extrapolated (by expanding the line at 0 Hz). In most cases though, DC conductivity cannot be compared with AC impedance. Table 3 presents the nanocomposite hydrogels studied by impedance spectroscopy.

Some hydrogel studies incorporate conductive materials which are not strictly defined as nanomaterials, *e.g.* hydrogels with conductive oligomers. We provide a short table of such hydrogels as ESI† (Table S1). It contains hydrogels with the incorporation of aniline oligomers, hybrid hydrogels with polypyrrole polymer incorporation and metallic microwires.

The nanomaterial concentrations in hydrogels can be described in many ways:  $\text{mg ml}^{-1}$ , % w/w, w/v, v/v on the total



Fig. 13 Conductive atomic force microscopy (C-AFM) and impedance spectroscopy of methacryloyl-substituted tropoelastin (MeTro) – rGO hydrogel. Surface topography and spatial conductivity of (A) pristine and (B) nanocomposite hydrogel. The nanomaterial affected both the topography and the electrical conductivity of the hydrogel. Current spikes are visible on the nanomaterials' locations. (C). Impedance spectroscopy of pristine (green) and nanocomposite hydrogel (red). Reduced GO decreased impedance in all of the frequency range studied. Adapted with permission from ref. 142. Copyright 2015 Wiley.





Table 2 Electrical conductivity of nanocomposite hydrogels

Nanomaterial	Polymer matrix	Application	Nanomaterial conc.	Hydrogel cond. $\sigma_0$ ( $\text{S m}^{-1}$ )	$\sigma_{\text{max}}$ ( $\text{S m}^{-1}$ )	Notes	Ref.
CNT	OPF oligo(poly(ethylene glycol)fumarate)	Neural tissue engineering	10 mg ml <sup>-1</sup>	$3.7 \times 10^{-4}$ (at 1 mg ml <sup>-1</sup> CNT)	$2.04 \times 10^{-3}$		72
CNT	PAM/PAA polyacrylamide/polyacrylic acid	Biosignals/skin protection	0.36% w/v	0.25	8	Glycerol + water hydrogels. CNTs functionalized with dopamine	146
CNT	Gelatin	Skin treatment	0.4% w/w	$2.5 \times 10^{-2}$	$7.2 \times 10^{-2}$	CNTs functionalized with dopamine	147
CNT (SW)	Gelatin	Muscle tissue engineering	1.5 mg ml <sup>-1</sup>	$3 \times 10^{-8}$	$5 \times 10^{-5}$	Gelatin cross-linked with GA	73
CNT (SW)	Collagen type I	Tissue engineering	4% w/w polymer	0.3	0.7	Glutaraldehyde CNTs carboxylated. Collagen seeded with HDF human dermal fibroblast cells	113
CNT (SW)	Collagen type I	Neural tissue engineering	0.01% w/w	1.4	2.4	CNTs oxidized. Measured cond. in prepolymer solution form	139
CNT (SW)	Collagen type I	Muscle tissue engineering	2% w/w polymer	0.25	0.61		148
CNT (SW)	Agarose	Motion sensor	15% w/w polymer	0.1 (at 5% CNT w/w of polymer)	0.3	CNTs functionalized with dopamine	145
CNT (DW)	Agarose	Drug delivery	40% w/w polymer	$7.4 \times 10^{-4}$	$2.7 \times 10^{-2}$		112
CNT (MW)	Agarose	Neural tissue engineering	0.1% w/w	0.94	1.46	Hydrogels lyophilized and then placed in a PBS solution before conductivity measurements	149
CNT (MW)	Gellan gum	Tissue engineering	0.1–0.5% w/v increased with drying	$10^{-3}$	$10^{-2}$ , up to 0.53 with decreasing water content		109
CNT (MW)	Gelatin	Drug delivery	8 mg ml <sup>-1</sup>	$1.3 \times 10^{-11}$	$2.6 \times 10^{-11}$		74
CNT (MW)	Alginate	Probe microbe electro-activity				Percolation at 0.5% w/w	110
CNT (MW)	PNIPAM poly <i>N</i> -isopropyl acrylamide	Motion sensor	4 mg ml <sup>-1</sup>		0.13	CNTs carboxylated. LAPO-NITE <sup>®</sup> nanoclay added for mechanical strength	150
CNT (MW)	PAM/CS polyacrylamide/chitosan	Motion sensor	0.125% w/v	0.1	0.95	CNTs carboxylated	44
CNT (MW)	PAM polyacrylamide	Artificial skin, Electromagnetic interference shielding	1% w/w	0.1	0.85	CNTs dispersed with cellulose nanofiber	143
CNT (MW)	poly(EA-MAA) ethyl acrylate – methacrylic acid		1% w/w	$4 \times 10^{-2}$	3		111
CNT (MW)	PEG polyethylene glycol	Neural tissue engineering	0.1% w/v	1.14	1.6		151
CNT (MW)	PEG polyethylene glycol	Neural tissue engineering	1.2% w/v	1.14	2		152
CNT (MW)	PEG/DFA polyethylene glycol/dimer fatty acid	Tissue engineering	6% w/w polymer	$2.4 \times 10$	$1.6 \times 10^{-2}$	CNTs oxidized	153
CNT (MW) + GO	OPF oligo(poly(ethylene glycol) fumarate)	Neural tissue engineering	10 mg ml <sup>-1</sup> rGO 10 mg ml <sup>-1</sup> CNT	$2 \times 10^{-4}$	$7.9 \times 10^{-4}$	CNTs functionalized with PEGA poly(ethylene glycol)acrylate	39
Carbon black	PGMA poly(glycerol methacrylate)	Bioelectronics	0.005% w/w	$10^{-4}$	0.1		154
Carbon black	PEGMA poly(ethylene glycol) dimethacrylate	Bioelectronics	0.005% w/w	$9.3 \times 10^{-5}$	$1.8 \times 10^{-2}$		154
Carbon black	PDEGMA poly(diethylene glycol) dimethacrylate	Bioelectronics	0.005% w/w	$8.9 \times 10^{-5}$	$1.2 \times 10^{-2}$		154
Carbon nanofibers	Chitosan	Muscle tissue engineering	1% w/v	0.03	0.04		155
GO	Chitosan	Tissue engineering	0.05% w/w	$5.7 \times 10^{-2}$	$1.22 \times 10^{-1}$	GO functionalized with dopamine	119
GO	Chitosan	Neural tissue engineering	3% w/w polymer	$10^{-8}$	$4 \times 10^{-2}$		156
GO	Chitosan	Tissue engineering	3% w/w polymer	$10^{-8}$	0.13	GO reduced	115



Table 2 (continued)

Nanomaterial	Polymer matrix	Application	Nanomaterial conc.	Hydrogel cond. $\sigma_0$ ( $\text{S m}^{-1}$ )	$\sigma_{\text{max}}$ ( $\text{S m}^{-1}$ )	Notes	Ref.
GO	PVA/PEG poly(vinyl alcohol)/polyethylene glycol	ECG acquisition	1.2% w/w	$8.5 \times 10^{-4}$	$5.1 \times 10^{-3}$		144
GO + PEDOT:PSS	PU polyurethane	Neural tissue engineering	14% w/w GO + PEDOT:PSS	0.13 (at 2% w/w GO + PEDOT:PSS)	0.62	Liquid crystals of reduced GO + PEDOT:PSS	157
GO	OPF oligo(poly(ethylene glycol) fumarate)	Muscle tissue engineering	1 mg $\text{ml}^{-1}$	0.09	0.42		158
GO	PVA poly(vinyl alcohol)		0.5% w/w PVA	$10^{-4}$	0.27	GO reduced and functionalized with dopamine.	79
GO	Sodium polyacrylate		10% w/w polymer	$5.9 \times 10^{-4}$	$3.7 \times 10^{-3}$	GO reduced	159
GO	PNIPAM poly <i>N</i> -isopropyl acrylamide		0.23% v/v	0.7 (at 0.05% v/v, 100 Hz)	9 (at 100 Hz)	A reduced GO aerogel (monolith) is mixed with the polymer	114
GO	PAM polyacrylamide	Muscle tissue engineering	0.3% w/v	$1.8 \times 10^{-3}$	$1.3 \times 10^{-2}$	GO reduced	160
GO	PAM polyacrylamide		4% w/w polymer	$2.5 \times 10^{-2}$	10	GO reduced and functionalized with dopamine	161
Graphene	PAM polyacrylamide		2% w/w polymer	–	$10^{-6}$		162
Graphene	PAA poly(acrylic acid)		1.25% v/v	$3 \times 10^{-14}$	$1.6 \times 10^{-5}$		80
Ag nanowires	Agarose	Motion sensor	50 mg $\text{ml}^{-1}$	0.01 (at 10 mg $\text{ml}^{-1}$ Ag nanowires)	300	Patterned silver nanowires	163
Ag	Collagen	Tissue engineering	$2.7 \times 10^{-7}$ % w/w	$4 \times 10^{-7}$	$8 \times 10^{-7}$		164
Au	Collagen	Tissue engineering	$4.9 \times 10^{-7}$ % w/w	$4 \times 10^{-7}$	$1.3 \times 10^{-6}$		164
Au	Chitosan	Muscle tissue engineering	1.5 w/w polymer	0.08	0.13		91
PANI nanofibers	PAA poly(acrylic acid)	Muscle tissue engineering	4.65% w/w		0.14	Doped with CSA camphor sulphonic acid	122
PANI	PVP poly( <i>N</i> -vinyl-2-pyrrolidone)		0.06% w/w	$5 \times 10^{-4}$	$10^{-3}$		138
PEDOT:PSS	GelMA gelatin methacrylate	Motion sensor	0.5% w/v	0.03	0.09	LAPONITE <sup>®</sup> nanoplatelets added for mechanical strength	87
PPy	GelMA gelatin methacrylate + PEGDA poly(ethylene glycol) diacrylate	Muscle tissue engineering	4 mg $\text{ml}^{-1}$	$3.6 \times 10^{-3}$ (at 1.2 $\times 10^{-2}$ 1 mg $\text{ml}^{-1}$ Ppy)			88

mass of the nanocomposite hydrogel or as a ratio over the polymer concentration, for example 5% CNT w/w of gelatin. Nanomaterial concentration expressed as a ratio over the total hydrogel mass or volume (water + polymer + nanomaterial) is more relevant for the electrical properties of the system (electrical percolation, for example, depends on the % v/v concentration) but does not remain stable over different conditions, since the water content of hydrogels may change due to evaporation or swelling. Nanomaterial concentration expressed as a ratio over the polymer concentration has the advantage of remaining stable on different conditions but is less relevant for the properties of the system as a whole. In the following tables, we have chosen to keep the nanomaterial concentration as reported by the original authors, due to a lack of sufficient data that would allow us to homogenize the concentrations (notably polymer concentrations and/or water content).

## 6.2 Processing methods to increase electrical conductivity

**6.2.1 Nanomaterial alignment.** Nanomaterials with high aspect ratios, such as nanotubes, nanosheets and nanowires can be aligned within the polymer matrix to further increase

the conductivity in the alignment orientation. The conductive nanomaterial alignment introduces an anisotropic conductivity to the hydrogel, a property which can be exploited in specific applications such as biosensing and bioelectronics.<sup>170,171</sup> Oriented nanomaterials on scaffolds for tissue engineering can also allow to direct the growth and differentiation of skeletal cells.<sup>171,172</sup> For example, mesenchymal stem cells on a substrate with aligned CNTs stretch along the direction of the CNTs. According to Namgung *et al.* the elongated stem cells have higher cytoskeletal tension which triggers mechanotransduction pathways translating into increased proliferation and differentiation when compared with cells growing on a substrate with randomly distributed CNTs.<sup>173</sup> Nanomaterials embedded in hydrogels can be aligned through mechanical strain,<sup>174</sup> with electrospinning<sup>175</sup> or through the application of an external AC electric field, a technique known as dielectrophoresis.<sup>165,176</sup>

**6.2.2 Dielectrophoresis.** In dielectrophoresis, an AC electric field exerts a force on dielectric particles, due to the charge polarisation on the particles and the surrounding medium.<sup>165</sup> It can be used in hydrogel nanoengineering, to orientate nanomaterials within a polymer matrix.



Table 3 Nanocomposite hydrogels characterized by impedance spectroscopy

Nanomaterial	Polymer matrix	Application	Nanomaterial conc.	Conductivity	Notes	Ref.
CNT	GelMA gelatin methacrylate	Muscle tissue engineering	0.3 mg ml <sup>-1</sup>	Impedance decreased with increased CNT content. Even more for aligned CNTs	CNTs carboxylated and aligned with dielectrophoresis	165
CNT (MW)	GelMA gelatin methacrylate	Muscle tissue engineering, actuator	5 mg ml <sup>-1</sup>	Increasing CNT content decreases impedance	CNTs carboxylated	166
CNT (MW)	GelMA gelatin methacrylate	Muscle tissue engineering	1 mg ml <sup>-1</sup>	Impedance decreased with increased CNT content. Aligned CNTs decreased impedance even at lower concentration (0.25 mg ml <sup>-1</sup> )	CNTs carboxylated and aligned with dielectrophoresis	167
GO	GelMA gelatin methacrylate	Tissue engineering	2 mg ml <sup>-1</sup>	1 mg ml <sup>-1</sup> GO did not affect impedance. 2 mg ml <sup>-1</sup> GO decreased impedance		168
GO	Methacryloyl-substituted tropoelastin	Muscle tissue engineering	2 mg ml <sup>-1</sup>	GO decreased impedance. rGO even more		142
Au nanorods	GelMA gelatin methacrylate	Muscle tissue engineering	1.5 mg ml <sup>-1</sup>	Impedance decreased with increasing gold nanorod content	Gold nanorods aspect ratio 3.15	36
PEDOT:PSS	PHEA poly(N-hydroxyethyl acrylamide)	Muscle tissue engineering	0.3% w/v	Impedance decreased with increasing PEDOT:PSS content		169

Small and Paunov elaborated an agarose – silver nanowire composite gel with anisotropic electrical conductivity.<sup>60</sup> They dissolved 0.5% w/v agarose in hot water (70 °C) and added 0.5% w/v 50 nm long silver nanowires. An AC electric field of 50 to 300 V cm<sup>-1</sup> and frequency of 5 kHz was applied to assemble and align the nanowires into percolating microwires in the direction of the applied field, while the temperature was kept at 55 °C. When the silver nanowires had aligned, the temperature was lowered to allow the agarose to form a gel, thus entrapping the nanowires. In a subsequent study by the same authors, the silver nanowires were functionalised with an antibody (thiolated biotin) and the electrically anisotropic hydrogel was used as a proof of concept for biosignal detection (Fig. 14(A)). An antigen (streptavidin) that binds on the antibody, induces tighter packing of the silver nanowires and limits their Brownian motion leading to an increased conductivity.<sup>60</sup>

Ramón-Azcón *et al.* prepared a gelatin methacrylate hydrogel with 0.3 mg ml<sup>-1</sup> CNTs. An AC electric field of 20 V<sub>pp</sub> and frequency of 2 MHz was applied to dielectrophoretically align the CNTs (Fig. 14(B)). The gelatin methacrylate was polymerised after aligning the CNTs, thus freezing their orientation.<sup>165</sup> In further work, they showed how the nanocomposite hydrogel with aligned CNTs enhanced the cardiac differentiation of mouse embryoid bodies. An externally applied electric pulse stimulation (1 V, 10 ms, 1 Hz) significantly increased the beating frequency of embryoid bodies on the nanocomposite hydrogels with aligned CNTs, as compared to the pristine hydrogels or the nanocomposite hydrogels with randomly distributed CNTs.<sup>167</sup>

**6.2.3 Mussel-inspired dopamine-coating of nanomaterials.** Efficient dispersion of nanomaterials is essential for their property-enhancing features. In particular, a substantial increase in electrical conductivity of nanocomposite hydrogels can only take place once the filler particles have formed a percolative network, and the agglomeration of nanomaterials dramatically increases the percolation threshold concentration.

Polydopamine coating is inspired by adhesive proteins secreted by mussels to attach to wet surfaces. Dopamine is a simple

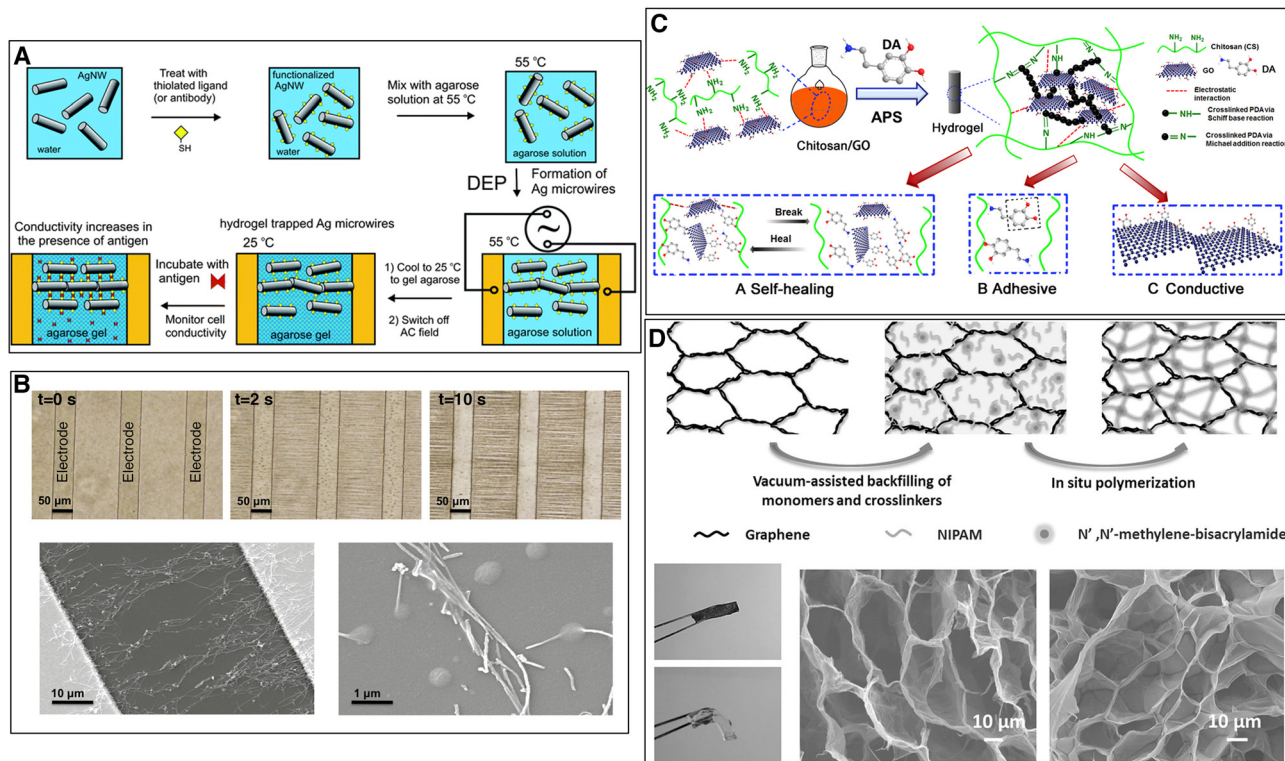
structural mimic of *Mytilus Edulis* foot protein 5 (Mefp-5).<sup>177</sup> In an alkaline solution, it self-polymerizes into a thin surface-adherent polydopamine film through oxidation by dissolved oxygen.<sup>147,178</sup> Polydopamine adheres to a large number of varied surfaces through covalent (Michael reaction of catechol with an amine or thiol) and non-covalent (hydrogen bonds and  $\pi$ - $\pi$  stacking) interactions.<sup>178</sup> In nanocomposite hydrogels, polydopamine coating is used to facilitate nanomaterial dispersion and to confer self-healing and adhesive properties to the hydrogel.<sup>119,161</sup>

Han *et al.* exploited the surface properties of polydopamine to fabricate a conductive and adhesive nanocomposite hydrogel. They coated CNTs with polydopamine and noted that the coating facilitated the dispersion of CNTs in an aqueous suspension. Next, they dissolved acrylamide and acrylic acid monomers in the suspension, before adding glycerol to form a glycerol–water binary solvent that allows the gel to be manipulated in a very wide range of temperatures (–20 °C to 60 °C). The glycerol–water gel was then formed *in situ* by UV-initiation of polyacrylamide/polyacrylic acid copolymerization. The gels with polydopamine coated CNTs showed a significant increase in electrical conductivity compared to gels without dopamine (from 5 to 8.2 S m<sup>-1</sup> for hydrogels with 10% w/w of polymer CNTs). The resulting hydrogel can find applications in adhesive bioelectronics to detect biosignals.<sup>146</sup>

Jing *et al.* conceived a self-healing, adhesive and conductive nanocomposite hydrogel for cardiac tissue engineering. They used a chitosan polymer, reinforced with 0.5 mg ml<sup>-1</sup> GO. Dopamine was added to cross-link the chitosan gel, partially reduce the graphene oxide (the oxidative polymerization of dopamine releases electrons) and facilitate its dispersion by coating it (Fig. 14(C)). The nanocomposite hydrogel showed a twofold increase in electrical conductivity (from 0.06 S m<sup>-1</sup> to 0.12 S m<sup>-1</sup> for chitosan-dopamine hydrogels) and enhanced the viability and proliferation of human stem cell-derived cardiomyocytes.<sup>119</sup> Zhang *et al.* drew on the previous work and proposed a flexible, wearable sensor consisting of a poly(vinyl alcohol) hydrogel matrix embedded with GO. Dopamine was







**Fig. 14** Processing methods to increase electrical conductivity of nanocomposite hydrogels. (A) Functionalized silver nanowires in warm water–agarose solution aligned through dielectrophoresis (DEP). After cooling down, agarose gelation occurs and a hydrogel with anisotropic electrical conductivity is formed. The conductivity increases in the presence of antigen due to the tighter packing of the functionalized silver nanowires. Reproduced from ref. 60 with permission from The Royal Society of Chemistry. (B) Nanocomposite hydrogel with chitosan, polydopamine and reduced GO. Self-healing mechanism attributable to the electrostatic interactions between GO and chitosan and the non-covalent bonding of catechol groups. Adhesiveness as a result of the catechol groups of polydopamine. Electrical conductivity through the reduced GO. Reprinted from ref. 119, Copyright 2016, with permission from Elsevier. Dielectrophoretic alignment of CNT between electrode bands. Phase images of CNT dielectrophoretic alignment over time. SEM images of dielectrophoretically aligned CNTs between two electrodes. Reprinted from ref. 167, Copyright 2017, with permission from Elsevier. (D) Graphene monolith – poly-*N*-isopropylacrylamide hydrogel. Photos of nanocomposite (dark and rigid) and pristine (transparent) hydrogel and SEM images of graphene monolith and freeze-dried nanocomposite hydrogel. Reprinted with permission from ref. 114. Copyright 2014 Wiley.

added to reduce and facilitate the dispersion of GO and the resulting hydrogel had a conductivity of  $0.5 \text{ S m}^{-1}$ . Small-scale movements of the human body (from joint movements to breathing) applied strain to the hydrogel, changing its electrical conductivity.<sup>179</sup>

Other applications of mussel-inspired conductive hydrogels include a patch to repair myocardial infarction,<sup>88</sup> wound dressing to promote the regeneration of infected skin,<sup>147,180</sup> a self-rolling hydrogel film biosensor<sup>181</sup> and an implantable biosensor for health monitoring.<sup>161</sup>

**6.2.4 Monolith (aerogel) conductive network.** Qiu *et al.* fabricated a cork-like, mechanically stable, elastic, graphene monolith (or aerogel) by freeze casting.<sup>182</sup> In following work, they developed a conductive hydrogel by inserting an aqueous solution of *N*-isopropylacrylamide monomers and polymerizing *in situ* (Fig. 14(D)). In contrast to conventional graphene nanocomposite hydrogels, where the percolation threshold is between 0.2 and 2% v/v, they reported a very low percolation threshold of 0.045% v/v. Additionally, no surface modification of graphene is required with this technique. Conventional nanocomposite hydrogels have a random distribution of

nanomaterials which also tend to form aggregates. The graphene monolith hydrogel was polymerized within an already percolated structure of graphene sheets.<sup>114</sup>

**6.2.5 Limitations of processing methods.** The above processing methods can increase the electrical conductivity of nanocomposite hydrogels by aligning the nanomaterials, improving their dispersion, and polymerizing a hydrogel within an already formed conductive network of nanomaterials. However, they do have some practical limitations. Alignment is only relevant for nanomaterials with high aspect ratios. It produces anisotropic conductivity, increasing the electrical conductivity in the direction of alignment and decreasing it in the perpendicular direction.<sup>171</sup> This may prove useful in some applications but disadvantageous for others. Polydopamine coating improves dispersion of hydrophobic nanomaterials but coating the surface of conductive nanomaterials with a polymer also decreases the electron tunnelling conductance. Lastly, not all nanomaterials can form self-standing 3D structures. CNTs and graphene are known to form monoliths which can then be vacuum-filled with a monomer solution and polymerized *in situ*. This may also be possible with metal nanowires but not with



spherical metal nanoparticles or conducting polymer nanomaterials. In addition, the precursor solution must have adequately low viscosity to enter the porous monoliths.

### 6.3 Comments and comparisons of electrical conductivities

Electrical conductivity of nanocomposite hydrogels varies by several orders of magnitude. The conductivity vs. nanomaterial concentration graph (Fig. 15) brings together the DC electrical conductivity values of nanocomposite hydrogels vs. nanomaterial concentration. It was compiled from the available literature data,<sup>44,72–74,87,88,91,109,111–115,119,144,146,147,151,152,155–160,164,183</sup> when enough information was available. Some observations can be made from these graphs and Table 2.

Nanomaterial concentrations employed to increase the conductivity of hydrogels were mostly within the range of 0.01 to 1% w/w. Some higher concentrations seen in the graph represent studies where nanocomposite hydrogels were partially dried, thus importantly increasing nanomaterial concentration.

Nanocomposite hydrogels can be classified as semi-conductors with their electrical conductivity spanning 6 orders of magnitude, from  $10^{-5}$  to  $10$  S m<sup>-1</sup>. The conductivities of hydrogels of different polymers, without nanomaterials, already show an important variability. We can distinguish two main groups. Firstly, the more insulating hydrogels, with conductivities of  $10^{-6}$  up to  $2 \times 10^{-3}$  S m<sup>-1</sup>. These are hydrogels made from agarose, alginate, hyaluronan, poly(vinyl alcohol), oligo(poly(ethylene glycol) fumarate) and polyacrylic acid. A second group of hydrogels have higher conductivities, even without nanomaterials, with values ranging from  $2 \times 10^{-2}$  up to  $1.4$  S m<sup>-1</sup>. They are

made from polymers including chitosan, polyethylene glycol, collagen, gelatin and poly(ethyl acrylate). Some of these differences (though not all) can be explained on the basis of polymer material chemistry. Polymers with ionisable functional groups (like amines and carboxylic acids) will confer a higher ionic conductivity to the hydrogel and polymers with conjugated chains have free electrons that contribute to the electronic conductivity of the polymer. Different measuring methods and conditions also contribute to the observed variability. A third group, not presented here, concerns hydrogels of conducting polymers. Conducting polymers include PPy, PANI, polythiophene, aniline oligomers and PEDOT. The hydrogels formed from these polymers are conductive but are brittle, have poor mechanical properties and are not biodegradable.<sup>57</sup>

The majority of nanocomposite hydrogels use carbon-based nanoparticles. GRMs and CNTs combine superior mechanical strength and electrical conductivity with high aspect ratios, leading to low percolation thresholds. By plotting the conductivity vs. concentration of these two nanomaterials in separate graphs, a different story emerges. GRMs show a relatively linear relationship between concentration and conductivity, on a log-log graph (Fig. 15(C)). Higher concentrations lead to higher conductivities. In the case of CNTs, there is no dominant trend on the graph (Fig. 15(B)). This is accentuated by some hydrogels that have high conductivity even without nanomaterial percolation (collagen and polyethylene glycol) but even by removing them, the relationship between concentration and conductivity is less straightforward.

Our hypothesis is that CNT agglomeration partly explains this difference. CNTs are highly hydrophobic and tend to form



**Fig. 15** Nanocomposite hydrogel electrical conductivity vs. nanomaterial concentration. (A) All, (B) CNTs, (C) GRMs. A nanomaterial concentration of 0 represents the electrical conductivity of the pristine hydrogels (without conductive nanomaterials). OPF: oligo(poly(ethylene glycol) fumarate); PAM: polyacrylamide; PEG: polyethylene glycol; PU: polyurethane. Data extracted from ref. 44, 72–74, 87, 88, 91, 109, 111–115, 119, 144, 146, 147, 151, 152, 155–160, 164 and 183.



agglomerates in aqueous suspensions. Hydrogels where the CNTs form bundles instead of uniformly dispersing in the hydrogel matrix do not form a percolation network, even in high concentrations. At the same time, CNTs have high aspect ratio (approximately 1000 or higher) and can, on the principle, form a percolating network even at low concentrations, if they are well dispersed. We maintain that, concerning the electrical conductivity of CNT-loaded hydrogels, dispersion is a parameter at least as important as concentration, if not more. In GRMs, hydrogels with rGO dominate the graph. GO is well dispersed in aqueous suspensions. The hydroxyl and carboxylic/carboxylate groups are hydrophilic, facilitating GO dispersion. GO has a much lower conductivity than graphene, but GO reduction can restore part of it. Better dispersions lead to a straightforward relationship between concentration and conductivity. We did not find in literature a sufficient number of hydrogels with polymer and metal nanomaterials to extend this reasoning.

## 7. Conclusion – perspectives

Nanocomposite hydrogels are complex, polyphasic materials, consisting of at least three components: water, a polymer matrix and the nanomaterials. In most cases additional components are also present: hybrid hydrogels with a second polymer, cross-linkers, dispersants and other impurities. All these components contribute to the electrical properties of the hydrogel through adding charge carriers (ions and electrons) and restricting or facilitating their mobility.

Moreover, nanocomposite hydrogels are dynamic systems. In aqueous environments they reach a dynamic equilibrium and their mass remains constant. When removed from an aqueous environment, and if the atmosphere is not saturated with water vapours, hydrogels lose over time a part of their water content. The kinetics of drying depend on the nature of the polymer, the nanomaterials and the environmental conditions (temperature, humidity), but can be quite rapid for most hydrogels even in ambient conditions. Electrical characterization equipment is mostly adapted to either solid materials or liquids. Hydrogels are viscoelastic materials with a high water content, falling between these two categories. The water loss affects electrical conductivity in a number of ways. Water loss changes the organisation of the hydrogel, densifying the polymer matrix. This directly reduces ion mobility, decreasing ionic conductivity. At the same time, the concentration of all components increases, due to the water loss. The nanomaterial configuration changes and can lead to a more connected network, increasing electronic conductivity. An example of this complexity is that despite all the efforts that have been undertaken, the percolation studies for nanocomposite hydrogels fail to reproduce the clear cut-off values found in studies of solid nanocomposite materials.

Hydrogels find numerous applications in biomedicine thanks to their biocompatibility, high water content and resemblance to native tissues. Materials scientists have been working on tailoring their properties, such as mechanical strength, water absorption capacity, porosity, adhesiveness and self-healability

to custom fit specific applications. In parallel, hydrogels can be made stimuli-responsive, reacting to environmental cues like temperature, pH, antigen presence, electromagnetic fields, and more. The incorporation of conductive nanomaterials can confer electrical conductivity to—otherwise insulating—hydrogels, expanding their field of application. These nanocomposite conductive hydrogels are widely used in tissue engineering, strain sensors and drug delivery where their electrical properties play a crucial role. Their applications will continue to expand as we further understand the link between the materials, the processing methods and their electrical properties.

## Abbreviations

CNT	Carbon nanotubes
SWNT	Single-wall carbon nanotubes
MWNT	Multi-wall carbon nanotubes
GRM	Graphene-related materials
GO	Graphene oxide
rGO	Reduced graphene oxide
PPy	Polypyrrole
PANI	Polyaniline
PEDOT	Poly-(3,4-ethylenedioxythiophene)
PSS	Poly(sodium-4-styrenesulfonate)

## Author contributions

G. Kougkoulos: writing – original draft; M. Golzio: writing – review & editing; L. Laudebat: writing – review & editing, supervision; Z. Valdez-Nava: writing – review & editing, supervision; E. Flahaut: writing – review & editing, supervision, funding acquisition.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 M. J. Lewis, in *Physical Properties of Foods and Food Processing Systems*, ed. M. J. Lewis, Woodhead Publishing, 1996, pp. 137–166.
- 2 Y. S. Zhang and A. Khademhosseini, *Science*, 2017, **356**(6337), 1–10.
- 3 K. Varaprasad, G. M. Raghavendra, T. Jayaramudu, M. M. Yallapu and R. Sadiku, *Mater. Sci. Eng., C*, 2017, **79**, 958–971.
- 4 F. Ullah, M. B. H. Othman, F. Javed, Z. Ahmad and H. Md. Akil, *Mater. Sci. Eng., C*, 2015, **57**, 414–433.





- 5 N. A. Peppas and A. S. Hoffman, in *Biomaterials Science*, ed. W. R. Wagner, S. E. Sakiyama-Elbert, G. Zhang and M. J. Yaszemski, Academic Press, 4th edn, 2020, pp. 153–166.
- 6 W. Wagermaier and P. Fratzl, in *Polymer Science: A Comprehensive Reference*, ed. K. Matyjaszewski and M. Möller, Elsevier, Amsterdam, 2012, pp. 35–55.
- 7 E. M. Ahmed, *J. Adv. Res.*, 2015, **6**, 105–121.
- 8 J. P. Gong, *Soft Matter*, 2006, **2**, 544–552.
- 9 H. Li, Y. S. Choi, M. W. Rutland and R. Atkin, *J. Colloid Interface Sci.*, 2020, **563**, 347–353.
- 10 A. S. Hoffman, *Adv. Drug Delivery Rev.*, 2012, **64**, 18–23.
- 11 A. Doderio, L. Pianella, S. Vicini, M. Alloisio, M. Ottonelli and M. Castellano, *Eur. Polym. J.*, 2019, **118**, 586–594.
- 12 S. J. Buwalda, K. W. M. Boere, P. J. Dijkstra, J. Feijen, T. Vermonden and W. E. Hennink, *J. Controlled Release*, 2014, **190**, 254–273.
- 13 D. Zhang, B. Ren, Y. Zhang, L. Xu, Q. Huang, Y. He, X. Li, J. Wu, J. Yang, Q. Chen, Y. Chang and J. Zheng, *J. Mater. Chem. B*, 2020, **8**, 3171–3191.
- 14 G. Isapour and M. Lattuada, *Adv. Mater.*, 2018, **30**, 1707069.
- 15 I. Tokarev, V. Gopishetty, J. Zhou, M. Pita, M. Motornov, E. Katz and S. Minko, *ACS Appl. Mater. Interfaces*, 2009, **1**, 532–536.
- 16 T. Distler and A. R. Boccaccini, *Acta Biomater.*, 2020, **101**, 1–13.
- 17 N. Chirani, L. Yahia, L. Gritsch, F. L. Motta, S. Chirani and S. Faré, *J. Biomed. Sci.*, 2016, **4**(2.13), 1–23.
- 18 B. W. Walker, R. P. Lara, E. Mogadam, C. H. Yu, W. Kimball and N. Annabi, *Prog. Polym. Sci.*, 2019, **92**, 135–157.
- 19 J. Stejskal, *Chem. Pap.*, 2017, **71**, 269–291.
- 20 K. Liu, S. Wei, L. Song, H. Liu and T. Wang, *J. Agric. Food Chem.*, 2020, **68**, 7269–7280.
- 21 A. Malti, J. Edberg, H. Granberg, Z. U. Khan, J. W. Andreasen, X. Liu, D. Zhao, H. Zhang, Y. Yao, J. W. Brill, I. Engquist, M. Fahlman, L. Wågberg, X. Crispin and M. Berggren, *Adv. Sci.*, 2016, **3**, 1500305.
- 22 B. Guo and P. X. Ma, *Biomacromolecules*, 2018, **19**, 1764–1782.
- 23 Q. Peng, J. Chen, T. Wang, X. Peng, J. Liu, X. Wang, J. Wang and H. Zeng, *InfoMat*, 2020, **2**, 843–865.
- 24 D. Liang, G. Zhou, Y. Hu, C. Zhao and C. Chen, *J. Mater. Sci.*, 2021, **56**, 14531–14541.
- 25 X. Zhang, Y. Tang, P. Wang, Y. Wang, T. Wu, T. Li, S. Huang, J. Zhang, H. Wang, S. Ma, L. Wang and W. Xu, *New J. Chem.*, 2022, **46**, 13838–13855.
- 26 C.-J. Lee, H. Wu, Y. Hu, M. Young, H. Wang, D. Lynch, F. Xu, H. Cong and G. Cheng, *ACS Appl. Mater. Interfaces*, 2018, **10**, 5845–5852.
- 27 M. Hess, R. G. Jones, J. Kahovec, T. Kitayama, P. Kratochvíl, P. Kubisa, W. Mormann, R. F. T. Stepto, D. Tabak, J. Vohlídal and E. S. Wilks, *Pure Appl. Chem.*, 2006, **78**, 2067–2074.
- 28 J. Cardoso, A. Huanosta and O. Manero, *Macromolecules*, 1991, **24**, 2890–2895.
- 29 C.-G. Wang, N. E. B. Surat'man, J. J. Chang, Z. L. Ong, B. Li, X. Fan, X. J. Loh and Z. Li, *Chem. – Asian J.*, 2022, **17**, e202200604.
- 30 S. Liu, J. Tang, F. Ji, W. Lin and S. Chen, *Gels*, 2022, **8**, 46.
- 31 A. Khademhosseini and R. Langer, *Nat. Protoc.*, 2016, **11**, 1775–1781.
- 32 J. A. Hunt, R. Chen, T. van Veen and N. Bryan, *J. Mater. Chem. B*, 2014, **2**, 5319–5338.
- 33 Y. Luo, in *Principles of Tissue Engineering*, ed. R. Lanza, R. Langer, J. P. Vacanti and A. Atala, Academic Press, 5th edn, 2020, pp. 343–360.
- 34 M. Petreaca and M. Martins-Green, in *Principles of Tissue Engineering*, ed. R. Lanza, R. Langer, J. P. Vacanti and A. Atala, Academic Press, 5th edn, 2020, pp. 93–117.
- 35 R. Dong, P. X. Ma and B. Guo, *Biomaterials*, 2020, **229**, 119584.
- 36 A. Navaei, H. Saini, W. Christenson, R. T. Sullivan, R. Ros and M. Nikkhah, *Acta Biomater.*, 2016, **41**, 133–146.
- 37 Y. Li, L. Wei, L. Lan, Y. Gao, Q. Zhang, H. Dawit, J. Mao, L. Guo, L. Shen and L. Wang, *Acta Biomater.*, 2022, **139**, 157–178.
- 38 P. Zarrintaj, A. M. Urbanska, S. S. Gholizadeh, V. Goodarzi, M. R. Saeb and M. Mozafari, *J. Colloid Interface Sci.*, 2018, **516**, 57–66.
- 39 X. Liu, A. L. M. Ii, S. Park, B. E. Waletzki, A. Terzic, M. J. Yaszemski and L. Lu, *J. Mater. Chem. B*, 2016, **4**, 6930–6941.
- 40 R. Eivazzadeh-Keihan, A. Maleki, M. de la Guardia, M. S. Bani, K. K. Chenab, P. Pashazadeh-Panahi, B. Baradaran, A. Mokhtarzadeh and M. R. Hamblin, *J. Adv. Res.*, 2019, **18**, 185–201.
- 41 J. Pelto, M. Björninen, A. Pälli, E. Talvitie, J. Hyttinen, B. Mannerström, R. Suuronen Seppanen, M. Kellomäki, S. Miettinen and S. Haimi, *Tissue Eng., Part A*, 2012, **19**, 882–892.
- 42 L. Tang, S. Wu, J. Qu, L. Gong and J. Tang, *Materials*, 2020, **13**, 3947.
- 43 Z. Chen, Y. Chen, M. S. Hedenqvist, C. Chen, C. Cai, H. Li, H. Liu and J. Fu, *J. Mater. Chem. B*, 2021, **9**, 2561–2583.
- 44 S. Xia, S. Song, F. Jia and G. Gao, *J. Mater. Chem. B*, 2019, **7**, 4638–4648.
- 45 M. Liao, P. Wan, J. Wen, M. Gong, X. Wu, Y. Wang, R. Shi and L. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1703852.
- 46 P. He, J. Wu, X. Pan, L. Chen, K. Liu, H. Gao, H. Wu, S. Cao, L. Huang and Y. Ni, *J. Mater. Chem. A*, 2020, **8**, 3109–3118.
- 47 H. Yuk, B. Lu and X. Zhao, *Chem. Soc. Rev.*, 2019, **48**, 1642–1667.
- 48 J. Li and D. J. Mooney, *Nat. Rev. Mater.*, 2016, **1**, 1–17.
- 49 H.-W. Liu, S.-H. Hu, Y.-W. Chen and S.-Y. Chen, *J. Mater. Chem.*, 2012, **22**, 17311–17320.
- 50 K. Elkhoury, P. Koçak, A. Kang, E. Arab-Tehrany, J. Ellis Ward and S. R. Shin, *Pharmaceutics*, 2020, **12**, 849.
- 51 S. Merino, C. Martín, K. Kostarelos, M. Prato and E. Vázquez, *ACS Nano*, 2015, **9**, 4686–4697.
- 52 S. Murdan, *J. Controlled Release*, 2003, **92**, 1–17.
- 53 A. Servant, V. Leon, D. Jasim, L. Methven, P. Limousin, E. V. Fernandez-Pacheco, M. Prato and K. Kostarelos, *Adv. Healthcare Mater.*, 2014, **3**, 1334–1343.
- 54 J. H. Lee, Y. C. Yoon, H. S. Kim, J. Lee, E. Kim, C. Findekle and U. Katscher, *Sci. Rep.*, 2022, **12**, 73.



- 55 F. A. Duck, in *Physical Properties of Tissues*, ed. F. A. Duck, Academic Press, London, 1990, pp. 167–223.
- 56 T. W. Balmer, S. Veszteg, P. Broekmann, A. Stahel and P. Büchler, *Sci. Rep.*, 2018, **8**, 8601.
- 57 A. Saberi, F. Jabbari, P. Zarrintaj, M. R. Saeb and M. Mozafari, *Biomolecules*, 2019, **9**, 448.
- 58 M. Rouabhia, H. Park, S. Meng, H. Derbali and Z. Zhang, *PLoS One*, 2013, **8**, e71660.
- 59 S. Du, N. Zhou, Y. Gao, G. Xie, H. Du, H. Jiang, L. Zhang, J. Tao and J. Zhu, *Nano Res.*, 2020, **13**, 2525–2533.
- 60 W. R. Small and V. N. Paunov, *J. Mater. Chem. B*, 2013, **1**, 5798–5805.
- 61 J.-F. Guillet, E. Flahaut and M. Golzio, *ChemPhysChem*, 2017, **18**, 2715–2723.
- 62 J. Simon, B. Jouanmiquieu, M.-P. Rols, E. Flahaut and M. Golzio, *Pharmaceutics*, 2021, **13**, 1805.
- 63 G. Kougkoulos, presented in part at SEEDS/JCGE22, Le Croisic, France, June 2022.
- 64 ISO/TR 18401:2017(en), Nanotechnologies—Plain language explanation of selected terms from the ISO/IEC 80004 series, <https://www.iso.org/obp/ui/#iso:std:iso:tr:18401:ed-1:v1:en>, (accessed 4 July 2022).
- 65 K. Zhou, *Carbon Nanomaterials: Modeling, Design, and Applications*, CRC Press, 2019.
- 66 M. Monthieux, P. Serp, E. Flahaut, M. Razafinimanana, C. Laurent, A. Peigney, W. Bacsá and J.-M. Broto, in *Springer Handbook of Nanotechnology*, ed. B. Bhushan, Springer, Berlin, Heidelberg, 2010, pp. 47–118.
- 67 M. F. L. D. Volder, S. H. Tawfick, R. H. Baughman and A. J. Hart, *Science*, 2013, **339**, 535–539.
- 68 J. Simon, E. Flahaut and M. Golzio, *Materials*, 2019, **12**, 624.
- 69 C.-W. Nan, Y. Shen and J. Ma, *Annu. Rev. Mater. Res.*, 2010, **40**, 131–151.
- 70 T. W. Ebbesen, H. J. Lezec, H. Hiura, J. W. Bennett, H. F. Ghaemi and T. Thio, *Nature*, 1996, **382**, 54–56.
- 71 F. Xie, P. Weiss, O. Chauvet, J. Le Bideau and J. F. Tassin, *J. Mater. Sci. Mater. Med.*, 2010, **21**, 1163–1168.
- 72 X. Liu, J. C. Kim, A. Lee Miller, B. E. Waletzki and L. Lu, *New J. Chem.*, 2018, **42**, 17671–17681.
- 73 J. Zhou, J. Chen, H. Sun, X. Qiu, Y. Mou, Z. Liu, Y. Zhao, X. Li, Y. Han, C. Duan, R. Tang, C. Wang, W. Zhong, J. Liu, Y. Luo, M. (Mengqiu) Xing and C. Wang, *Sci. Rep.*, 2014, **4**, 3733.
- 74 U. G. Spizzirri, S. Hampel, G. Cirillo, F. P. Nicoletta, A. Hassan, O. Vittorio, N. Picci and F. Iemma, *Int. J. Pharm.*, 2013, **448**, 115–122.
- 75 X. Du, I. Skachko, A. Barker and E. Y. Andrei, *Nat. Nanotechnol.*, 2008, **3**, 491–495.
- 76 P. Wick, A. E. Louw-Gaume, M. Kucki, H. F. Krug, K. Kostarelos, B. Fadeel, K. A. Dawson, A. Salvati, E. Vázquez, L. Ballerini, M. Tretiach, F. Benfenati, E. Flahaut, L. Gauthier, M. Prato and A. Bianco, *Angew. Chem., Int. Ed.*, 2014, **53**, 7714–7718.
- 77 J. Chen, B. Yao, C. Li and G. Shi, *Carbon*, 2013, **64**, 225–229.
- 78 A. J. Marsden, D. G. Papageorgiou, C. Vallés, A. Liscio, V. Palermo, M. A. Bissett, R. J. Young and I. A. Kinloch, *2D Mater.*, 2018, **5**, 032003.
- 79 M. Wang, Y. Chen, R. Khan, H. Liu, C. Chen, T. Chen, R. Zhang and H. Li, *Colloids Surf., A*, 2019, **567**, 139–149.
- 80 A. Alam, Q. Meng, G. Shi, S. Arabi, J. Ma, N. Zhao and H.-C. Kuan, *Compos. Sci. Technol.*, 2016, **127**, 119–126.
- 81 W. Humphrey, A. Dalke and K. Schulten, *J. Mol. Graphics*, 1996, **14**, 33–38.
- 82 K. Momma and F. Izumi, *J. Appl. Crystallogr.*, 2011, **44**, 1272–1276.
- 83 R. Ravichandran, S. Sundarajan, J. R. Venugopal, S. Mukherjee and S. Ramakrishna, *J. R. Soc., Interface*, 2010, **7**, S559–S579.
- 84 J. Pecher and S. Mecking, *Chem. Rev.*, 2010, **110**, 6260–6279.
- 85 J. H. Min, M. Patel and W.-G. Koh, *Polymers*, 2018, **10**, 1078.
- 86 Y. Bu, H.-X. Xu, X. Li, W.-J. Xu, Y. Yin, H. Dai, X. Wang, Z.-J. Huang and P.-H. Xu, *RSC Adv.*, 2018, **8**, 10806–10817.
- 87 Y. Li, X. Xiong, X. Yu, X. Sun, J. Yang, L. Zhu, G. Qin, Y. Dai and Q. Chen, *Polym. Test.*, 2019, **75**, 38–47.
- 88 L. Wang, J. Jiang, W. Hua, A. Darabi, X. Song, C. Song, W. Zhong, M. M. Q. Xing and X. Qiu, *Adv. Funct. Mater.*, 2016, **26**, 4293–4305.
- 89 P. Zhang, I. Wyman, J. Hu, S. Lin, Z. Zhong, Y. Tu, Z. Huang and Y. Wei, *Mater. Sci. Eng., B*, 2017, **223**, 1–23.
- 90 T. Bruna, F. Maldonado-Bravo, P. Jara and N. Caro, *Int. J. Mol. Sci.*, 2021, **22**, 7202.
- 91 P. Baei, S. Jalili-Firoozinezhad, S. Rajabi-Zeleti, M. Tafazzoli-Shadpour, H. Baharvand and N. Aghdami, *Mater. Sci. Eng., C*, 2016, **63**, 131–141.
- 92 M. Naguib, V. N. Mochalin, M. W. Barsoum and Y. Gogotsi, *Adv. Mater.*, 2014, **26**, 992–1005.
- 93 J.-C. Lei, X. Zhang and Z. Zhou, *Front. Phys.*, 2015, **10**, 276–286.
- 94 Y.-Z. Zhang, J. K. El-Demellawi, Q. Jiang, G. Ge, H. Liang, K. Lee, X. Dong and H. N. Alshareef, *Chem. Soc. Rev.*, 2020, **49**, 7229–7251.
- 95 A. S. Zeraati, S. A. Mirkhani, P. Sun, M. Naguib, P. V. Braun and U. Sundararaj, *Nanoscale*, 2021, **13**, 3572–3580.
- 96 Y. Zhang, M. Gong and P. Wan, *Matter*, 2021, **4**, 2655–2658.
- 97 A. Bhat, S. Anwer, K. S. Bhat, M. I. H. Mohideen, K. Liao and A. Qurashi, *npj 2D Mater. Appl.*, 2021, **5**, 1–21.
- 98 A. Aharony and D. Stauffer, *Introduction To Percolation Theory: Second Edition*, Taylor & Francis, 2018.
- 99 J. W. Essam, *Rep. Prog. Phys.*, 1980, **43**, 833.
- 100 W. Bauhofer and J. Z. Kovacs, *Compos. Sci. Technol.*, 2009, **69**, 1486–1498.
- 101 L. Berhan and A. M. Sastry, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2007, **75**, 041120.
- 102 J. R. Winkler and H. B. Gray, *J. Am. Chem. Soc.*, 2014, **136**, 2930–2939.
- 103 A. P. Chatterjee, *J. Phys.: Condens. Matter*, 2015, **27**, 315303.
- 104 A. P. Chatterjee, *J. Chem. Phys.*, 2013, **139**, 224904.
- 105 G. Ambrosetti, C. Grimaldi, I. Balberg, T. Maeder, A. Danani and P. Ryser, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 155434.
- 106 V. Ambegaokar, B. I. Halperin and J. S. Langer, *Phys. Rev. B: Solid State*, 1971, **4**, 2612–2620.



- 107 L. Berhan and A. M. Sastry, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2007, **75**, 041121.
- 108 H.-B. Zhang, W.-G. Zheng, Q. Yan, Y. Yang, J.-W. Wang, Z.-H. Lu, G.-Y. Ji and Z.-Z. Yu, *Polymer*, 2010, **51**, 1191–1196.
- 109 C. John Ferris and M. in het Panhuis, *Soft Matter*, 2009, **5**, 3430–3437.
- 110 L. Mottet, D. L. Cornec, J.-M. Noël, F. Kanoufi, B. Delord, P. Poulin, J. Bibette and N. Bremond, *Soft Matter*, 2018, **14**, 1434–1441.
- 111 Z. Cui, M. Zhou, P. J. Greensmith, W. Wang, J. A. Hoyland, I. A. Kinloch, T. Freemont and B. R. Saunders, *Soft Matter*, 2016, **12**, 4142–4153.
- 112 J.-F. Guillet, Z. Valdez-Nava, M. Golzio and E. Flahaut, *Carbon*, 2019, **146**, 542–548.
- 113 R. A. MacDonald, C. M. Voge, M. Kariolis and J. P. Stegemann, *Acta Biomater.*, 2008, **4**, 1583–1592.
- 114 L. Qiu, D. Liu, Y. Wang, C. Cheng, K. Zhou, J. Ding, V.-T. Truong and D. Li, *Adv. Mater.*, 2014, **26**, 3333–3337.
- 115 S. Sayyar, E. Murray, B. C. Thompson, J. Chung, D. L. Officer, S. Gambhir, G. M. Spinks and G. G. Wallace, *J. Mater. Chem. B*, 2014, **3**, 481–490.
- 116 T. Chen, K. Hou, Q. Ren, G. Chen, P. Wei and M. Zhu, *Macromol. Rapid Commun.*, 2018, **39**, 1800337.
- 117 P. Thoniyot, M. J. Tan, A. A. Karim, D. J. Young and X. J. Loh, *Adv. Sci.*, 2015, **2**, 1400010.
- 118 A. Hajian, S. B. Lindström, T. Pettersson, M. M. Hamed and L. Wågberg, *Nano Lett.*, 2017, **17**, 1439–1447.
- 119 X. Jing, H.-Y. Mi, B. N. Napiwocki, X.-F. Peng and L.-S. Turng, *Carbon*, 2017, **125**, 557–570.
- 120 B. Lu, H. Yuk, S. Lin, N. Jian, K. Qu, J. Xu and X. Zhao, *Nat. Commun.*, 2019, **10**, 1043.
- 121 J. Alam, L. A. Dass, M. S. Alhoshan, M. Ghasemi and A. W. Mohammad, *Appl. Water Sci.*, 2012, **2**, 37–46.
- 122 S. Hosseinzadeh, S. M. Rezayat, E. Vasheghani-Farahani, M. Mahmoudifard, S. Zamanlui and M. Soleimani, *Polymer*, 2016, **97**, 205–216.
- 123 A. Kisiel, D. Korol, A. Michalska and K. Maksymiuk, *Electrochim. Acta*, 2021, **390**, 138787.
- 124 M. Pishvaei, F. Najafi and M. Salami-Kalajahi, *J. Color Sci. Technol.*, 2016, **10**, 145–154.
- 125 C. Kittel, *Introduction to Solid State Physics*, Wiley, 8th edn, 2005.
- 126 F.-C. Chiu, *Adv. Mater. Sci. Eng.*, 2014, **204**, e578168.
- 127 S. M. Sze and K. K. Ng, *Physics of Semiconductor Devices*, Wiley, 3rd edn, 2006.
- 128 H. L. Tuller and P. K. Moon, *Mater. Sci. Eng., B*, 1988, **1**, 171–191.
- 129 C. Lin and I. Gitsov, *Macromolecules*, 2010, **43**, 3256–3267.
- 130 V. M. Gun'ko, I. N. Savina and S. V. Mikhalovsky, *Gels*, 2017, **3**, 37.
- 131 T. Hatakeyama and F. X. Quinn, *Thermal analysis: fundamentals and applications to polymer science*, Wiley, Chichester, New York, 2nd edn, 1999.
- 132 A. R. Khare and N. A. Peppas, *Polymer*, 1993, **34**, 4736–4739.
- 133 H. B. Lee, M. S. Jhon and J. D. Andrade, *J. Colloid Interface Sci.*, 1975, **51**, 225–231.
- 134 J. R. Macdonald and W. B. Johnson, *Impedance Spectroscopy*, John Wiley & Sons, Ltd, 2018, pp. 1–20.
- 135 N. Bonanos, B. C. H. Steele, E. P. Butler, J. R. Macdonald, W. B. Johnson, W. L. Worrell, G. A. Niklasson, S. Malmgren, M. Strømme, S. K. Sundaram, M. C. H. McKubre, D. D. Macdonald, G. R. Engelhardt, E. Barsoukov, B. E. Conway, W. G. Pell, N. Wagner, C. M. Roland and R. S. Eisenberg, *Impedance Spectroscopy*, John Wiley & Sons, Ltd, 2018, pp. 175–478.
- 136 P. Pissis and A. Kyritsis, *Solid State Ionics*, 1997, **97**, 105–113.
- 137 H. Warren, R. D. Gately, P. O'Brien, R. Gorkin and M. in het Panhuis, *J. Polym. Sci., Part B: Polym. Phys.*, 2014, **52**, 864–871.
- 138 C. Dispenza, C. L. Presti, C. Belfiore, G. Spadaro and S. Piazza, *Polymer*, 2006, **47**, 961–971.
- 139 A. N. Koppes, K. W. Keating, A. L. McGregor, R. A. Koppes, K. R. Kearns, A. M. Ziemba, C. A. McKay, J. M. Zuidema, C. J. Rivet, R. J. Gilbert and D. M. Thompson, *Acta Biomater.*, 2016, **39**, 34–43.
- 140 E. H. Hall, *Am. J. Math.*, 1879, **2**, 287–292.
- 141 G. Benstetter, A. Hofer, D. Liu, W. Frammelsberger and M. Lanza, *Conductive Atomic Force Microscopy*, John Wiley & Sons, Ltd, 2017, pp. 45–77.
- 142 N. Annabi, S. R. Shin, A. Tamayol, M. Miscuglio, M. A. Bakooshi, A. Assmann, P. Mostafalu, J.-Y. Sun, S. Mithieux, L. Cheung, X. (Shirley) Tang, A. S. Weiss and A. Khademhosseini, *Adv. Mater.*, 2016, **28**, 40–49.
- 143 W. Yang, B. Shao, T. Liu, Y. Zhang, R. Huang, F. Chen and Q. Fu, *ACS Appl. Mater. Interfaces*, 2018, **10**, 8245–8257.
- 144 X. Xiao, G. Wu, H. Zhou, K. Qian and J. Hu, *Polymers*, 2017, **9**, 259.
- 145 Q. Han, A. Wang, W. Song, M. Zhang, S. Wang, P. Ren, L. Hao, J. Yin and S. Bai, *ACS Appl. Bio Mater.*, 2021, **4**(8), 6148–6156.
- 146 L. Han, K. Liu, M. Wang, K. Wang, L. Fang, H. Chen, J. Zhou and X. Lu, *Adv. Funct. Mater.*, 2018, **28**, 1704195.
- 147 Y. Liang, X. Zhao, T. Hu, Y. Han and B. Guo, *J. Colloid Interface Sci.*, 2019, **556**, 514–528.
- 148 H. Sun, J. Zhou, Z. Huang, L. Qu, N. Lin, C. Liang, R. Dai, L. Tang and F. Tian, *Int. J. Nanomed.*, 2017, **12**, 3109–3120.
- 149 Z. Liu, M. Yushan, Y. Alike, Y. Liu, S. Wu, C. Ma and A. Yusufu, *BioMed Res. Int.*, 2020, **2020**, e4794982.
- 150 Z. Deng, T. Hu, Q. Lei, J. He, P. X. Ma and B. Guo, *ACS Appl. Mater. Interfaces*, 2019, **11**, 6796–6808.
- 151 M. Imaninezhad, K. Pemberton, F. Xu, K. Kalinowski, R. Bera and S. P. Zustiak, *J. Neural Eng.*, 2018, **15**, 056034.
- 152 K. Shah, D. Vasileva, A. Karadaghy and S. P. Zustiak, *J. Mater. Chem. B*, 2015, **3**, 7950–7962.
- 153 M. Mihajlovic, M. Mihajlovic, P. Y. W. Dankers, R. Masereeuw and R. P. Sijbesma, *Macromol. Biosci.*, 2019, **19**, 1800173.
- 154 P. Dutta, N. N. Dass and N. S. Sarma, *React. Funct. Polym.*, 2015, **90**, 25–35.
- 155 A. M. Martins, G. Eng, S. G. Caridade, J. F. Mano, R. L. Reis and G. Vunjak-Novakovic, *Biomacromolecules*, 2014, **15**, 635–643.





- 156 M. Jafarkhani, Z. Salehi and T. Nematian, *Mater. Today Proc.*, 2018, **5**, 15620–15628.
- 157 M. Javadi, Q. Gu, S. Naficy, S. Farajikhah, J. M. Crook, G. G. Wallace, S. Beirne and S. E. Moulton, *Macromol. Biosci.*, 2018, **18**, 1700270.
- 158 J. Zhou, X. Yang, W. Liu, C. Wang, Y. Shen, F. Zhang, H. Zhu, H. Sun, J. Chen, J. Lam, A. G. Mikos and C. Wang, *Theranostics*, 2018, **8**, 3317–3330.
- 159 R. Peng, Y. Yu, S. Chen, Y. Yang and Y. Tang, *RSC Adv.*, 2014, **4**, 35149–35155.
- 160 H. Jo, M. Sim, S. Kim, S. Yang, Y. Yoo, J.-H. Park, T. H. Yoon, M.-G. Kim and J. Y. Lee, *Acta Biomater.*, 2017, **48**, 100–109.
- 161 L. Han, X. Lu, M. Wang, D. Gan, W. Deng, K. Wang, L. Fang, K. Liu, C. W. Chan, Y. Tang, L.-T. Weng and H. Yuan, *Small*, 2017, **13**, 1601916.
- 162 S. Das, F. Irin, L. Ma, S. K. Bhattacharia, R. C. Hedden and M. J. Green, *ACS Appl. Mater. Interfaces*, 2013, **5**, 8633–8640.
- 163 Q. Han, Y. Chen, W. Song, M. Zhang, S. Wang, P. Ren, L. Hao, A. Wang, S. Bai and J. Yin, *Bio-Des. Manuf.*, 2019, **2**, 269–277.
- 164 K. Hosoyama, M. Ahumada, C. D. McTiernan, J. Bejjani, F. Variola, M. Ruel, B. Xu, W. Liang, E. J. Suuronen and E. I. Alarcon, *RSC Adv.*, 2017, **7**, 47704–47708.
- 165 J. Ramón-Azcón, S. Ahadian, M. Estili, X. Liang, S. Ostrovidov, H. Kaji, H. Shiku, M. Ramalingam, K. Nakajima, Y. Sakka, A. Khademhosseini and T. Matsue, *Adv. Mater.*, 2013, **25**, 4028–4034.
- 166 S. R. Shin, S. M. Jung, M. Zalabany, K. Kim, P. Zorlutuna, S. Bok Kim, M. Nikkhah, M. Khabiry, M. Azize, J. Kong, K. Wan, T. Palacios, M. R. Dokmeci, H. Bae, X. (Shirley) Tang and A. Khademhosseini, *ACS Nano*, 2013, **7**, 2369–2380.
- 167 S. Ahadian, S. Yamada, J. Ramón-Azcón, M. Estili, X. Liang, K. Nakajima, H. Shiku, A. Khademhosseini and T. Matsue, *Acta Biomater.*, 2016, **31**, 134–143.
- 168 S. R. Shin, B. Aghaei-Ghareh-Bolagh, T. T. Dang, S. N. Topkaya, X. Gao, S. Y. Yang, S. M. Jung, J. H. Oh, M. R. Dokmeci, X. (Shirley) Tang and A. Khademhosseini, *Adv. Mater.*, 2013, **25**, 6385–6391.
- 169 A. R. Spencer, A. Primbetova, A. N. Koppes, R. A. Koppes, H. Fenniri and N. Annabi, *ACS Biomater. Sci. Eng.*, 2018, **4**, 1558–1567.
- 170 S. J. Lee, G. L. Pishko, G. W. Astary, T. H. Mareci and M. Sarntinoranont, *J. Appl. Polym. Sci.*, 2009, **114**, 1992–2002.
- 171 S. Ahadian, J. Ramón-Azcón, M. Estili, X. Liang, S. Ostrovidov, H. Shiku, M. Ramalingam, K. Nakajima, Y. Sakka, H. Bae, T. Matsue and A. Khademhosseini, *Sci. Rep.*, 2014, **4**, 4271.
- 172 Z. J. Han, A. E. Rider, M. Ishaq, S. Kumar, A. Kondyurin, M. M. M. Bilek, I. Levchenko and K. (Ken) Ostrikov, *RSC Adv.*, 2013, **3**, 11058–11072.
- 173 S. Namgung, K. Y. Baik, J. Park and S. Hong, *ACS Nano*, 2011, **5**, 7383–7390.
- 174 C. M. Voge, M. Kariolis, R. A. MacDonald and J. P. Stegemann, *J. Biomed. Mater. Res., Part A*, 2008, **86A**, 269–277.
- 175 Y. Wu, L. Wang, B. Guo and P. X. Ma, *ACS Nano*, 2017, **11**, 5646–5659.
- 176 W. R. Small and V. N. Paunov, *J. Mater. Chem.*, 2008, **18**, 2082–2084.
- 177 H. Lee, S. M. Dellatore, W. M. Miller and P. B. Messersmith, *Science*, 2007, **318**, 426–430.
- 178 J. H. Ryu, P. B. Messersmith and H. Lee, *ACS Appl. Mater. Interfaces*, 2018, **10**, 7523–7540.
- 179 Y. Zhang, B. Liang, Q. Jiang, Y. Li, Y. Feng, L. Zhang, Y. Zhao and X. Xiong, *Smart Mater. Struct.*, 2020, **29**, 075027.
- 180 Y. Yang, Y. Liang, J. Chen, X. Duan and B. Guo, *Bioact. Mater.*, 2022, **8**, 341–354.
- 181 J. Jiang, Y. Huang, Y. Wang, H. Xu, M. Xing and W. Zhong, *Materials*, 2017, **10**, 964.
- 182 L. Qiu, J. Z. Liu, S. L. Y. Chang, Y. Wu and D. Li, *Nat. Commun.*, 2012, **3**, 1241.
- 183 J. W. Cho, J. W. Kim, Y. C. Jung and N. S. Goo, *Macromol. Rapid Commun.*, 2005, **26**, 412–416.

