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ARTICLE

Seed-mediated growth of MOF-encapsulated Pd@Ag core-shell nanoparticles: toward advanced room temperature nanocatalysts

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The possibility of using inner cavities within metal-organic frameworks (MOFs) as templates for the fabrication of tiny metal nanoparticles (NPs) was attempted in this work. An unprecedented design of Pd@Ag core-shell NPs on MOF via a seed mediated growth strategy is reported and attributed to the presence of activated physisorbed hydrogen atoms on embedded Pd NPs as reducing agents to selectively direct the deposition of Ag onto Pd while minimizing Ag self-nucleation. The obtained Pd@Ag core-shell NPs exhibited a significant increase in selectivity in the partial hydrogenation of phenylacetylene as compared to their monometallic counterparts, due to the surface dilution and electron modification of the surface Pd sites by Ag deposition. Pd@Ag NPs also possessed an unprecedented high stability and recyclability in the catalytic reactions, related to the nano-confinement effect and strong metal-support interaction offered by the MOF framework.

Introduction

Bimetallic core-shell nanoparticles (NPs) have attracted increasing attention in recent years because of their fascinating properties and applications in a variety of fields, especially in catalysis.¹ The inner core metal can significantly influence the external shell of another metal to provide unique physico-chemical properties.² These properties can be tuned by controlling the size, composition and structure of bimetallic core-shell NPs. In this regard, template-directed syntheses were demonstrated as effective strategies to control the growth of metal NPs by the pore size, shape and channel structure of porous materials (e.g., zeolites, mesoporous silicas and carbon nanotubes).³ Nevertheless, few attempts were reported to date on the preparation of ultrasmall bimetallic core-shell NPs within several nanometers using porous materials as templates.⁴

Metal-organic frameworks (MOFs) emerged as a new class of porous crystalline materials featuring tunable pore structures, large internal surface areas and promising functionalities.⁵ These properties make MOFs very appropriate candidates for the stabilization of metal NPs.⁶ The nodal metal ions and aromatic backbone of MOFs can particularly interact weakly with embedded metal NPs through coordination and π - π forces, which can lead to charge transfer to such NPs to trigger enhanced stability/activity performance.⁷ The possibility of using the inner cavities of MOFs as templates for the design of

tiny metal/metal oxides NPs can be therefore considered of high relevance in the design of advanced nanocatalysts.⁸ Previous studies proved the successful incorporation of monometallic or bimetallic alloy NPs within the pores of MOFs.⁹ However, the incorporation of bimetallic core-shell NPs into MOFs pores remains a great challenge as the core-shell NPs would be more easily deposited on the external surface of MOFs.^{4a-b} Such difficulties come from the inherent MOF microstructures which difficult structure and composition tailoring of embedded species in the restricted space. To the best of our knowledge, there is only one example to date claiming the preparation of Pd@Co core-shell NPs within MOF by a simultaneous reduction method.^{4c} Nevertheless, the simultaneous reduction method is not widely employed for core-shell NPs as it produces alloyed NPs in most cases.^{1b}

Seed-mediated growth has been well-documented as most powerful route among the developed synthetic methods to synthesize bimetallic core-shell NPs.¹⁰ In a typical seed mediated growth method, pre-formed seeds of one metal serve as nucleation sites for further growth of another metal. During subsequent reductive growth of the metal shell, a sophisticated and careful control is required to avoid individual nucleation and growth of the secondary metal as individual particles.

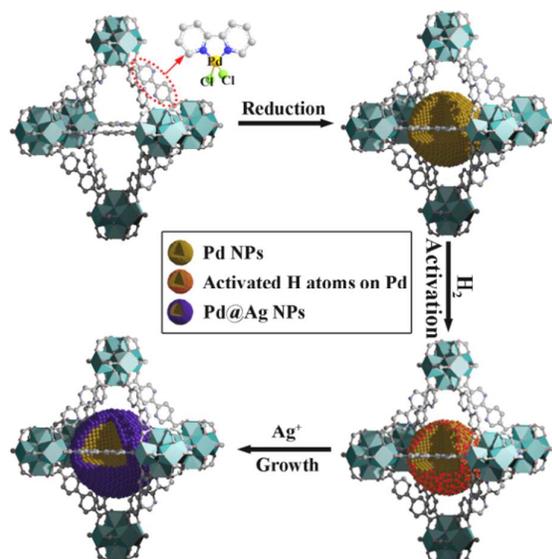
Herein, we report a simple, efficient and unprecedented approach for the preparation of ultrafine Pd@Ag core-shell NPs within the pores of MOF under a seed-mediated growth strategy with activated hydrogen atoms as reducing agent. This strategy involves a multistep 1) first encapsulation of Pd NPs within MOF through a pre-incorporation method followed by 2) dissociation and activation of hydrogen molecules on the surface of Pd NPs to serve as effective reducing agents for 3) a selective deposition of Ag on Pd (Scheme 1). The success of the

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proposed strategy can be ascribed to the activated hydrogen atoms confined on the embedded Pd NP surfaces, which promoted the exclusive reduction of Ag^+ on Pd and prevented a significant self-nucleation of Ag to generate individual Ag NPs. Such a rational design was demonstrated to provide encapsulated ultrafine Pd@Ag core-shell NPs with various Pd/Ag ratios and average sizes of ca. 2.6–3.1 nm within MOF pores. Interestingly, the Ag shell could effectively block the high coordination sites on the Pd core, thus leading to a significant increase in selectivity in the partial hydrogenation of phenylacetylene selected as model reaction.



Scheme 1. Schematic illustration of the fabrication of Pd@Ag core-shell NPs encapsulated in the MOF pore.

Results and discussion

UiO-67 MOF, based on $\text{Zr}_6\text{O}_4(\text{OH})_4(\text{CO}_2)_{12}$ secondary building units (SBUs) and dicarboxylate bridging ligands was selected as host matrix for the encapsulation of NPs due to their high physicochemical stability and tunable functionalities of ligands as well as large BET surface areas (up to $2000 \text{ m}^2/\text{g}$).¹¹ In our previous work, we have developed a pre-incorporation strategy to exclusively encapsulate Pd NPs within the pores of MOF, preventing any external surface deposition.^{9a} We also demonstrated that hydrogen could be dissociated into atomic hydrogen and spillover onto the surface of Pd- or Pt-MOF composites.¹² Highly active H atoms possessed a remarkably superior reducing ability as compared to H_2 to effectively reduce Ag^+ to metallic Ag otherwise unachievable using molecular hydrogen at room temperature.¹³

In a typical synthesis of Pd@Ag-in-UiO-67, Pd precursors were first incorporated within UiO-67 using a pre-incorporation strategy (see Supporting Information for details), followed by treatment under H_2 at 200°C for 2 h to yield Pd-in-UiO-67. Pd-in-UiO-67 was subsequently dispersed in DMF and the solution was bubbled with H_2 for 1 h at room temperature

followed by the addition of a AgNO_3 solution under stirring, to promote the growth of Ag on the surface of embedded Pd NPs.

As measured by atomic absorption spectroscopy (AAS), the Pd/Ag molar ratio in the as-prepared Pd@Ag-in-UiO-67 was calculated to be 1/1.4. Powder X-ray diffraction (PXRD) patterns of UiO-67 and metal-loaded materials are presented in Figure S1. The incorporation of metals in UiO-67 did not cause any apparent loss of crystallinity, indicating that the MOF structure was mostly preserved upon metal incorporation. No identifiable peaks for metal NPs were observed most probably due to the low metal content in the materials. Porosity is one of the key factors that determine the ability of porous materials for catalytic applications. N_2 adsorption/desorption experiments (Figure S2) before and after loading of metal NPs exhibited a mixture of type I and IV curves. Surface areas obtained were 2415, 2101, and $1893 \text{ m}^2 \text{ g}^{-1}$ for as-synthesized UiO-67, Pd-in-UiO-67, Pd@Ag-in-UiO-67 (1:1.4), respectively (Table S1). The observed lower surface areas of UiO-67-encapsulated metal samples with respect to those of the parent UiO-67 indicated that the cavities of UiO-67 could be partially occupied by metal NPs.

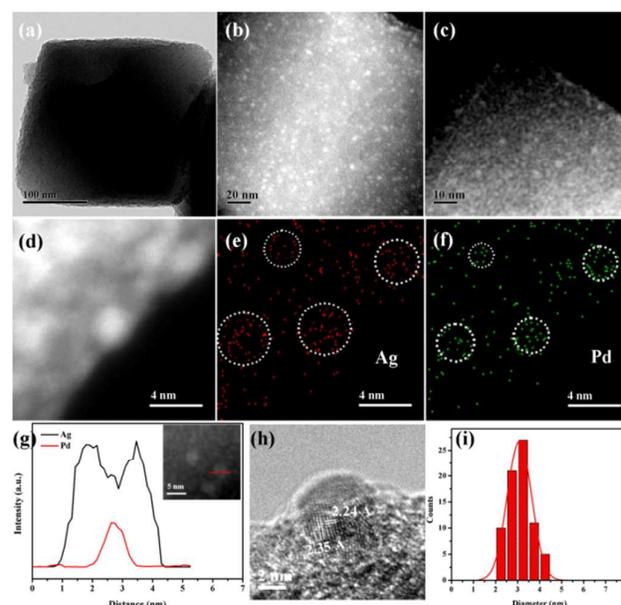


Figure 1. TEM image of Pd@Ag-in-UiO-67 (1:1.4) (a). HAADF-STEM image of ultrathin cuts from Pd@Ag-in-UiO-67 (1:1.4) (b). HAADF-STEM image of ultrathin cuts from Pd-in-UiO-67 (1:1.4) (c). HAADF-STEM image of ultrathin cuts from Pd@Ag-in-UiO-67 (1:1.4) (d) and the corresponding STEM-EDX elemental mapping images of Ag and Pd (e-f). Elemental line-scanning spectra of Pd@Ag-in-UiO-67 (1:1.4) (inset) along the direction marked by a red line which unambiguously confirmed the formation of core-shell NPs (g). HRTEM image of a Pd@Ag NP (h). The corresponding size distribution of Pd@Ag core-shell NPs (i).

The morphology and structure of Pd-in-UiO-67 and Pd@Ag-in-UiO-67 (1:1.4) were investigated by transmission electron microscopy (TEM), high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM), and energy-dispersive X-ray spectroscopy (EDS) elemental mapping analyses. Pd-in-UiO-67 contained a homogeneous distribution of Pd NPs within UiO-67, with an average size of 2.5 nm (Figure 1c). HAADF-STEM images of ultrathin slices from Pd-in-UiO-67

suggested that Pd NPs were mostly located inside UiO-67 (Figure S3). After Ag growth onto Pd NPs, the obtained Pd@Ag NPs possessed an average particle size of 3.1 ± 0.5 nm, featuring an excellent dispersion in UiO-67 with an octahedral shape (Figure 1a). Note that the particle sizes were somewhat exceeding the pore size of MOF. Such a phenomenon might be interpreted by the local defects/deformations of the host frameworks by the growth of metal NPs (MNPs), which is in accordance with other MNPs@MOFs composites.^{4a,9a} HAADF-STEM image of ultrathin micrometer-sized slices of Pd@Ag-in-UiO-67 further confirmed the presence of Pd@Ag NPs homogeneously distributed within the MOF framework (Figure 1c). Although the Pd@Ag core-shell structure was difficult to be distinguished from HAADF-STEM images due to the very close atom masses between Pd and Ag, the structure could be unambiguously demonstrated by EDS mapping and line profile analysis. As shown in Figure 1d-g, the element Pd was distributed only in the core with an Ag-rich shell, suggesting that the growth of Ag exclusively took place on the Pd surface. More importantly, no isolated Ag NPs could be visualized in the materials. A representative high-resolution TEM (HRTEM) image of a deliberately selected large Pd@Ag NP showed that the interplanar spacing of the particle lattice were 0.224 and 0.235 nm, corresponding to the (111) lattice spacing of face-centered cubic Pd and cubic Ag, respectively (Figure 1h).

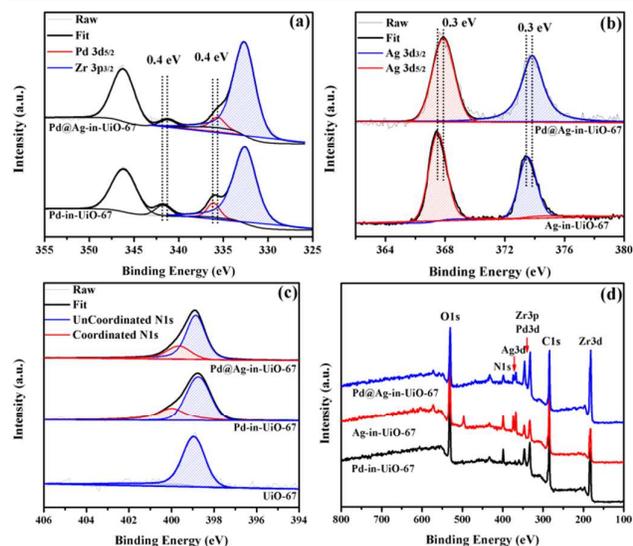


Figure 2. XPS spectra of Pd-in-UiO-67, Ag-in-UiO-67, and Pd@Ag-in-UiO-67 (1:1.4) referenced to hydrocarbon C 1s. a) Pd 3d, b) Ag 3d, c) N 1s, and d) survey photo emission spectrum.

X-Ray photoelectron spectroscopy (XPS) experiments with Ar etching were subsequently conducted to further confirm the structural and electronic environment of the core-shell NPs. XPS data showed that the Pd 3d 5/2 peak for Pd@Ag-in-UiO-67 (1:1.4) was located at around 335.7 eV, corresponding to zero-valent Pd species. As compared to monometallic Pd-in-UiO-67, the binding energy was interestingly shifted to a lower value by approximately 0.4 eV (Figure 2a). In the case of Pd@Ag-in-UiO-67, the observed Ag 3d 5/2 binding energy at

367.9 eV clearly corresponds to Ag⁰, observed at higher binding energies with respect to Ag-in-UiO-67 (Figure 2b). These results pointed to a modification of the electronic structure of Pd and Ag atoms, further confirming the formation of bimetallic NPs. XPS of Pd@Ag-in-UiO-67 before and after Ar etching also revealed an Ag-rich shell and a Pd core (Figure S4). Additionally, a new N 1s peak at ca. 400 eV was observed for UiO-67-encapsulated metal samples as compared to the N 1s peak in the pristine UiO-67 (Figure 2c). Such N1s shift towards higher binding energies was attributed to the slight transfer of electrons to the embedded metal NPs. Such an electron transfer observed from XPS spectra would support the fact that Pd@Ag NPs possessed certain chemical interactions with the MOF support.

The synthetic procedure could be readily extended to the preparation of Pd@Ag-in-UiO-67 materials with different Pd/Ag ratios. As Pd-in-UiO-67 was allowed to bubble with excess H₂, surface Pd atoms were expected to be (almost) fully covered by activated H atoms. Each H atom donated one electron for the reductive deposition of Ag⁺ on Pd. By varying the amount of AgNO₃ added into the synthesis solution, the Pd/Ag ratio in the obtained Pd@Ag-in-UiO-67 could be simply controlled until Pd was fully covered. Pd to Ag ratios in the obtained Pd@Ag NPs could be decreased from 30/1 to 1/1.4 when the Ag amount added to the synthesis system was increased from 30 μg to 2 mg as determined by AAS analysis (Table S2). Interestingly, a further increase in Ag amount could maintain 1/1.4 Pd/Ag ratios implying that Pd was fully covered by Ag. TEM images clearly proved that the average size of Pd@Ag NPs increased from 2.6 to 3.1 nm as the Pd/Ag ratio decreased from 30/1 to 1/1.4 (Figure S5). This indicated that Pd NPs were decorated with separated Ag atoms (at low Ag loadings) as the size of the obtained Pd@Ag was similar to that of pure Pd NPs. Ag atoms would then entirely cover the Pd surface at higher Ag loadings (e.g., Pd/Ag=1:1.4).

After detailed characterization, *as-prepared* Pd and Pd@Ag NPs confined in UiO-67 were explored as catalysts for the partial-hydrogenation of phenylacetylene. The catalytic partial-hydrogenation of alkynes to alkenes is one of the most important transformations in petrochemistry and industry.¹⁴ However, the challenge to achieve high selectivities to alkenes at high conversions (due to the easy over-hydrogenation of alkenes to undesired alkanes) still remains to date.

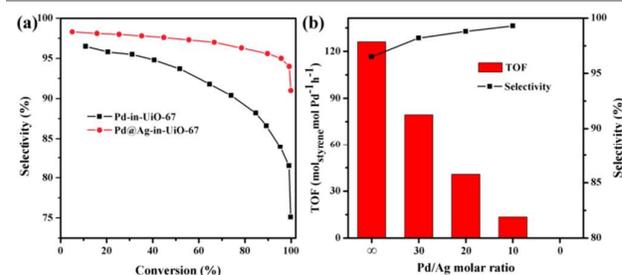


Figure 3. a) Selectivity to styrene as a function of phenylacetylene conversion for reaction over Pd-in-UiO-67 and Pd@Ag-in-UiO-67 (30:1). b) Plot of TOF value and selectivity to styrene (calculated at 10% conversion of phenylacetylene) versus Pd/Ag molar ratio in Pd@Ag-in-UiO-67. TOF: moles of phenylacetylene produced on per mole of Pd per hour.

All reactions were performed at room temperature and atmospheric hydrogen pressure. The influence of phenylacetylene conversion on styrene selectivity was monitored using Pd-in-UiO-67 and Pd@Ag-in-UiO-67 (30:1) catalysts. For both catalysts, styrene selectivity generally declined with an increase in phenylacetylene conversion. However, Pd@Ag-in-UiO-67 exhibited an apparently improved selectivity (91%, quantitative phenylacetylene conversion) as compared to monometallic Pd NPs confined in UiO-67 (75% selectivity, >99% phenylacetylene conversion, Figure 3a). Importantly, an increase in Ag loading led to a significant enhancement in styrene selectivity, accompanied by a gradual loss in activity (Figure 3b and Figure S6). With no catalytic activity observed for Ag-in-UiO-67 under the investigated reaction conditions, the observed decrease in activity after Ag deposition can be clearly correlated to the blocking of active surface Pd sites by a Ag shell, confirming the formation of Pd@Ag systems. Pd@Ag-in-UiO-67 (1:1.4) provided no catalytic activity under the investigated conditions, further confirming that the Pd surface was fully covered by Ag.

The enhanced selectivity of Pd@Ag-in-UiO-67 as compared to Pd-in-UiO-67 may be attributed to surface dilution and electron modification effects of Ag on the surface Pd sites. For monometallic Pd NPs, the presence of adjacent Pd sites in large Pd ensembles could simply favor a full hydrogenation of styrene to ethylbenzene. Interestingly, the presence of Ag in the Pd@Ag NPs can originate Pd surface dilution and result in isolated Pd sites, which could suppress over hydrogenation of phenylacetylene of crucial importance for the selective production of styrene.¹⁵ The noticeable improvements in selectivity can also be attributed to a modification of the electronic structure of surface Pd owing to its interaction with Ag, in which a charge transfer Ag-to-Pd was present (as demonstrated by XPS analysis). The formation of electronegative Pd species could lead to a decrease in the ratio of phenylacetylene/styrene adsorption energies, thus suppressing over-hydrogenation of phenylacetylene.^{15a}

Catalytic stability is of great importance for practical applications of highly efficient catalysts. After reaction completion, the catalyst was recovered from the solution and then reutilized for another reaction run under identical reaction conditions using fresh reagents. Pd@Ag-in-UiO-67 (30:1) showed no significant changes in terms of catalytic activity and selectivity after successive reuses up to five cycles (Figure 4a). The reused catalyst was also analyzed using PXRD, TEM and AAS to determine its structural and chemical stability. PXRD pattern and TEM of reused catalyst remained virtually unchanged in terms of structure, nanoparticle sizes and dispersion, confirming also the maintained structure of UiO-67 (Figure 4b-d).

The doped metal content of the reused material was also confirmed to be almost identical to the fresh catalyst. Leaching experiments were eventually performed to verify the key role of Pd@Ag NPs as catalytically active species as well as to confirm the heterogeneous nature of the catalyst. With this purpose, a typical reaction was conducted using Pd@Ag-in-UiO-67 (30:1) as catalyst during 15 min after which the catalyst was removed from the reaction mixture by centrifugation. Upon catalyst removal, the mixture in the absence of catalyst was left reacting for several hours without any further observed increase in conversion which strongly suggested that the reaction was heterogeneously catalyzed (Figure S7). These findings were consistent with AAS experiments as no Pd traces (below the detection limit, 0.5 ppm) were observed in the reaction mixture. These findings indicated that Pd@Ag-in-UiO-67 was a highly active and stable catalyst for room temperature hydrogenations under the investigation conditions, suggesting that the confinement effect offered by the framework and the strong metal-support interactions should play important roles in preventing Pd@Ag NPs from aggregation.

Conclusions

In summary, we have successfully fabricated ultrafine core-shell bimetallic Pd@Ag NPs within the pores of MOFs via a seed mediated growth strategy. The deposition of Ag on Pd could separate active Pd sites and modify the electronic structure of Pd, leading to a significant increase in selectivity in the room temperature partial-hydrogenation of phenylacetylene selected as model reaction. Interestingly, Ag/Pd ratios could be easily tuned in the synthesized Pd@Ag UiO-67 by controlling the addition of different amounts of AgNO₃. This effective and unprecedented strategy reported herein could facilitate the control of size, structure and composition of a range of metal NPs within the pores MOF pores for the potential design of encapsulated core-shell nanocomposites with applications in a variety of fields including heterogeneous catalysis, photocatalysis and biomass conversion to be further reported in due course.

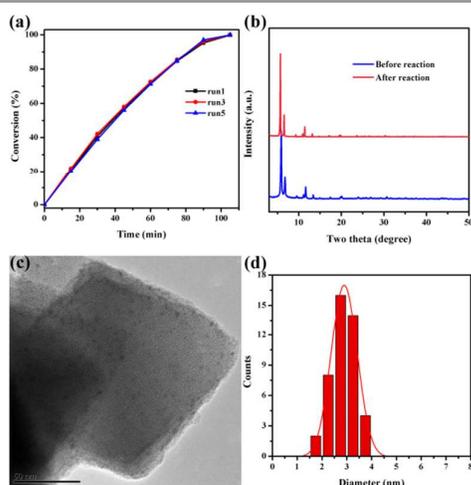


Figure 4. a) Reusability of Pd@Ag-in-UiO-67 (30:1). b) Powder XRD patterns of Pd@Ag-in-UiO-67 (30:1) before and after reaction. c) TEM image of Pd@Ag-in-UiO-67 (30:1) after being used for five times, (d) corresponding size distribution of Pd@Ag NPs.

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