

Cite this: *Dalton Trans.*, 2022, **51**, 7401

# Advances and perspectives of actinide chemistry from *ex situ* high pressure and high temperature chemical studies

Gabriel L. Murphy, \* Philip Kegler and Evgeny V. Alekseev \*

High pressure high temperature (HP/HT) studies of actinide compounds allow the chemistry and bonding of among the most exotic elements in the periodic table to be examined under the conditions often only found in the severest environments of nature. Peering into this realm of physical extremity, chemists have extracted detailed knowledge of the fundamental chemistry of actinide elements and how they contribute to bonding, structure formation and intricate properties in compounds under such conditions. The last decade has resulted in some of the most significant contributions to actinide chemical science and this holds true for *ex situ* chemical studies of actinides resulting from HP/HT conditions of over 1 GPa and elevated temperature. Often conducted in tandem with *ab initio* calculations, HP/HT studies of actinides have further helped guide and develop theoretical modelling approaches and uncovered associated difficulties. Accordingly, this perspective article is devoted to reviewing the latest advancements made in actinide HP/HT *ex situ* chemical studies over the last decade, the state-of-the-art, challenges and discussing potential future directions of the science. The discussion is given with emphasis on thorium and uranium compounds due to the prevalence of their investigation but also highlights some of the latest advancements in high pressure chemical studies of transuranium compounds. The perspective also describes technical aspects involved in HP/HT investigation of actinide compounds.

Received 4th March 2022,  
Accepted 20th April 2022

DOI: 10.1039/d2dt00697a

rsc.li/dalton

## Introduction

Since the dawn of the nuclear age, the chemistry of actinide elements and the compounds they form has captivated the attention of society, particularly regarding the deliberate or unintentional applications of their nuclear properties. Transcending this, considerable efforts have been made by chemists to understand the nature of chemical bonding within actinide materials and the associated properties that are derived therein. Pertinent to the chemistry of the actinides is the pronounced relativistic effects which their core electrons present and the occurrence of considerable spin-orbit coupling which vary significantly between individual elements. Although the theoretical framework for relativistic and heavy electron systems has been available since the early days of quantum theory,<sup>1</sup> accounting for these effects in *ab initio* wave-function based calculations has only recently become somewhat accessible.<sup>2</sup> However, as contemporary studies have highlighted<sup>3</sup> limitations still remain in executing correct standalone calculations without the aid of experimental data to benchmark obtained results.<sup>4–9</sup> Consequently, to correctly understand chemical bonding in actinide compounds requires controlled experimental approaches

that can both directly probe and examine their limitations in a systematic form. *Ex situ* HP/HT experimental methods have subsequently formed a powerful means to achieve this and advance fundamental chemical insight. Typically, published studies of actinides under high pressure and temperature are relatively few compared to those of ambient pressure elevated temperature studies, a consequence of the radiological hazard combined with operating costs and difficulties associated with HP/HT apparatuses. Nevertheless, those that are available have often divulged significant insight into the fundamental chemistry of actinides, pertinent in the context of basic scientific endeavour and as applied functional materials for both nuclear and non-nuclear applications.

In the early days of actinide high pressure science, studies were initially conducted using *ex situ* belt-type HP/HT apparatuses, which although can be considered relatively primitive by today's standards, hinted at the flexibility and peculiarity of actinide bonding, and could maintain conditions above 4 GPa and over 1000 °C. For instance, in 1966 Young<sup>10</sup> and shortly after in 1967 Hoekstra and Marshall<sup>11</sup> using such apparatuses with post powder diffraction measurements concluded the formation of NiUO<sub>4</sub> polymorphs could only be achieved through pressure induced overlap of atomic orbitals resulting in a distorted structure of which was later confirmed by single crystal X-ray diffraction methods.<sup>3,12</sup>

Institute of Energy and Climate Research, Forschungszentrum Jülich GmbH,  
52428 Jülich, Germany. E-mail: g.murphy@fz-juelich.de, e.alekseev@fz-juelich.de



The advent of the diamond anvil cell (DAC) by Van Valkenburg and co-workers<sup>13</sup> heralded a new era in high-pressure science, of which the actinides were not excluded. It meant smaller sample amounts could be examined *in situ* at greater pressures reducing radiological risk whilst dramatically increasing the operating range compared to that offered by piston cylinder or other multi-anvil devices. Accordingly, it resulted in a decline of studies using *ex situ* HP/HT apparatuses in place of DAC related *in situ* studies. The latter studies in the past were generally focused on the metallic form of actinides but also the binary oxides, due to their applications and relevance to the nuclear fuel cycle. Such studies led to the acquisition of the high-pressure phases of UO<sub>2</sub> and ThO<sub>2</sub>.<sup>14</sup> The discovery of heavy fermion systems in the late 1970s, invigorated fundamental interest by physicists of placing actinides under pressure in exploring the superconductive phases and condensed electron systems of the notable compounds URu<sub>2</sub>Si<sub>2</sub>, and UGe<sub>2</sub> among others.<sup>15,16</sup>

The turn of the century coincided with new trends in actinide chemical investigation where the utilisation of precise high-resolution measurements of compound arising from a multitude of conditions such as high temperature, hydrothermal, solvothermal and supercritical water conditions.<sup>17–21</sup> This resulted in one of greatest volumes of published literature for actinides of any period. These inorganic chemistry studies were conducted with the purpose of building a fundamental understanding of actinide chemistry behind the nuclear fuel cycle and particularly nuclear waste materials. However, a notable absentee of the conditions examined in this period was that of pressures above 1 GPa and under elevated temperatures. This limitation of conditions examined within previous studies, and by extension associated knowledge of actinide chemistry therein, is well defined observed within a *p*-*T*

diagram as is presented in Fig. 1. From the figure, it is apparent that the breadth of field for HP/HT conditions that actinides can be examined greatly exceeds that of those produced by high temperature, hydrothermal, solvothermal or supercritical water conditions. Although high temperature and hydrothermal methods can involve some elevated pressure, such as using sealed quartz tubes, it is dwarfed by what HP/HT techniques can access. Nevertheless, this realm of heightened pressure and temperature is greatly understudied.

Outside of the laboratory, actinides can be exposed to HP/HT conditions in a variety of real and engineering systems, in addition to nature. Terrestrially in nature, uranium and thorium are most abundant of the actinides (22 and 80 ppb for bulk silicate Earth) and are typically found within the Earth's crust (2.7 and 9.6 ppm) which extends to a depth of 70 km corresponding to pressures and temperatures of over 600 °C and >1–2 GPa.<sup>22,23</sup> Actinide occurrence within the Earth extends even further to within the Earth's mantle at depths of 2890 km where pressures and temperatures of over 135–140 GPa and 3650 °C can be encountered.<sup>23</sup> It is here actinides, particularly uranium and thorium, provide a key proportion of the radiogenic input into the Earth' internal heat budget.<sup>24</sup> Anthropogenically, actinides occur within and around applications of the nuclear fuel cycle and weapons technology. Within a light water reactor (LRW) the primary coolant circuit typically operates at temperatures of 315 °C with pressures around 10 MPa, whereas in future designs such as the Supercritical Water Reactor (SCWR), as part of GEN IV reactors, considers approximate pressures and temperatures of 25 MPa and 370 °C.<sup>25,26</sup> Within the centre of LWR UO<sub>2</sub> fuels, HP/HT conditions can be more realised as the centreline fuel pellet temperature is held typically below 1400 °C<sup>27</sup> and it has been estimated the internal pressures generated during He bubble

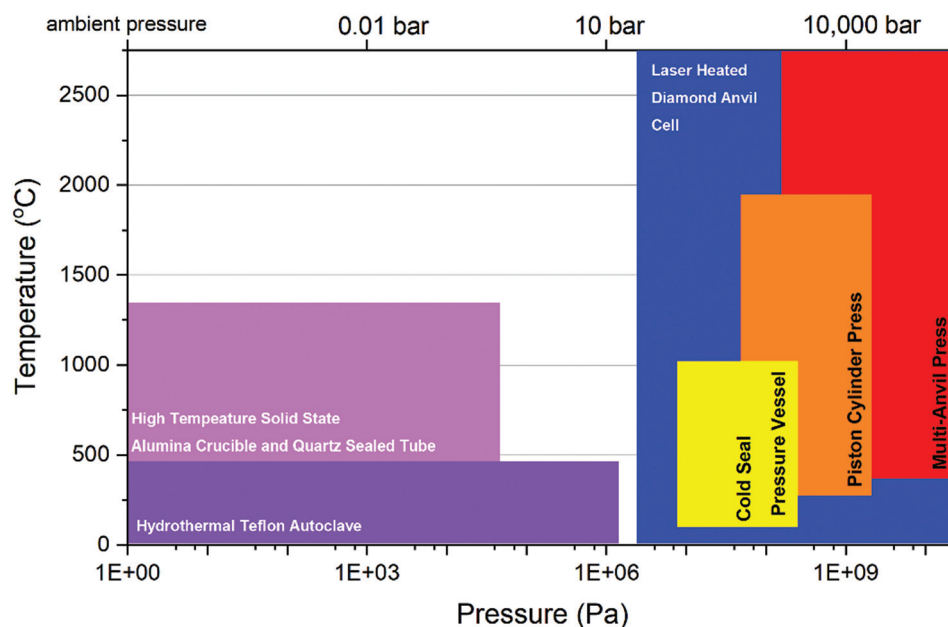


Fig. 1 Typical experimental methods used for examining the chemistry of actinides with respect to temperatures and pressures they can access.



formation can reach GPa level.<sup>28</sup> Within spent nuclear fuel (SNF) near-surface geological repositories, such as the planned Swedish Forsmark SNF repository, maximum pressures and temperatures of a few MPa and around 100 °C of these near surface facilities are predicted.<sup>29</sup> In deep borehole designs (DBD), which have been examined due favourable safety aspects, account for pressures of over 150 MPa and significantly higher temperatures than that of near surface facility types.<sup>30</sup>

As a waste stream from fuel enrichment, depleted uranium (DU) has found considerable recycled usage in military munitions due its high density and pyrophoricity. Depending on firing velocity, projectile design and target composition pressures and temperatures of over several MPa and above 600 °C can be reached in which oxidation of the DU metal rapidly occurs resulting in binary and ternary oxide formation such as U<sub>3</sub>O<sub>8</sub>, UO<sub>3</sub>, FeUO<sub>4</sub> and Fe<sub>2</sub>UO<sub>6</sub> from primary impact.<sup>31,32</sup> The synthesis of FeUO<sub>4</sub> is known to require elevated pressure. During the detonation of a nuclear weapon, depending on the design, efficiency and assembly of the warhead, extreme pressures and temperatures can be encountered. For instance, temperatures and pressures of over 7700 °C and above 8 GPa are suspected to have occurred during the Trinity test detonation of the 'gadget' in 1945.<sup>33</sup> From these extreme conditions, the peculiar and infamous mineral phase trinitite with some An-content was recovered, which formed through the combination of these extreme conditions with the nuclear debris and surrounding sand.<sup>34</sup> Although this description of where actinides can potentially encounter HP/HT conditions only scratches the surface of scenarios, it is evident w.r.t. Fig. 1, minimum hydrothermal and better yet, HP/HT pressure devices are necessary to appropriately uncover and explore the chemical properties that can occur in these often extreme places.

The Fukushima Daiichi nuclear reactor incident of 2011 led to the public perception of nuclear energy sinking to one of its lowest points. To ameliorate this, significant efforts were made by industry and governments to ensure the safety of reactor and nuclear systems, whilst behind this, actinide researchers began focusing on understanding the behaviour of actinides when exposed to extreme environments, for instance what is potentially encountered in an accident scenario. This culminated in the re-emergence of HP/HT experimental work, particularly of *ex situ* method, on actinides systems towards understanding their chemistry when placed in extreme scenarios and understanding changes to their bonding and properties therein. Subsequently, the previous decade afforded some of most significant output of studies relating to *ex situ* HP/HT investigation of actinides, particularly the early members, Th and U. These studies have shed significant light on their fundamental chemistry as compounds but also on the nature of their bonding. Considering this and that there are no current review or perspective articles available to this topic, the present perspective article is devoted to discussing the advances in actinide chemical science through HP/HT *ex situ* studies that use pressures greater than 1 GPa and under elevated temperatures over the last decade. Focus will be given primarily to the early actinides,

Th and U, due to the significant steps made in their research, particularly on the future direction of HP/HT *ex situ* science, but discussion also on recent advances on the later actinides. Note unless specially stated otherwise, *e.g. in situ* DAC related studies, *ex situ* HP/HT studies will be often referred to as HP/HT studies throughout the perspective article.

## High pressure/high temperature *ex situ* technical aspects

Placing actinides under HP/HT conditions in a laboratory setting is not a trivial task and despite many apparatuses available globally, few are licensed to run actinides due the radiological risk. Of the HP/HT actinide synthesis studies produced over the previous decade, the Voggenreiter LP 1000-540/50, a combined piston cylinder/multi anvil press (Fig. 2) located at the Institute of Reactor Safety and Nuclear Waste Management (IEK-6) Forschungszentrum Jülich Germany, has produced the majority of those available.

The Voggenreiter LP 1000-540/50 uses an end loaded piston cylinder module allowing pressures up to 4.5 GPa and temperatures up to 1800 °C, which can be easily replaced by a Walker-type multi anvil module allowing conditions up to 24 GPa and 2300 °C.<sup>35</sup> The combination of classical piston cylinder and multi anvil techniques with today's fully computerised temperature and pressure control,<sup>36</sup> allow precise experimental conditions to be maintained for the synthesis of comparably large quantities (depending on experimental pressure of up to hundreds of milligrams) of new HP/HT materials, allowing for



Fig. 2 The Voggenreiter LP 1000-540/50 HP/HT apparatus present in the laboratories of the Institute of Reactor Safety and Nuclear Waste Management (IEK-6), Forschungszentrum Jülich, Germany.



the *ex situ* measurements of the products in standard in house analytical devices like powder and single crystal diffraction. The complex assemblies for both modules, which are designed for the generation of hydrothermal pressure conditions, are internally heated (graphite or LaCrO<sub>3</sub> heater) and temperature control is conducted using an *in situ* thermocouple. A comparable setup was used by Huppertz, Hinteregger and co-workers<sup>37–40</sup> for their high-pressure research on 3d and 4f elements. Hydrothermal conditions ( $p < 0.25$  GPa,  $t < 850$  °C) with sample quantities of up to 1 gram can be achieved using a cold seal vessel<sup>41</sup> (here water is used as pressure media in an externally heated pressure vessel, often made out of inconel®). The sample itself is introduced into a capsule that is made from a noble metal (mostly gold) which is placed inside the pressure vessel, this is then closed using a water cooled, cone shaped seal. This method was used in synthesis of a uranyl aluminoborate complex.<sup>41</sup>

## Thorium HP/HT studies

Interest in thorium chemistry has typically been overshadowed by that of its more societally relevant sister, uranium. The further absence of 5f electrons and preference to remain tetravalent typically leads to a perception that thorium is less exotic and more mundane chemically than uranium. However as contemporary investigations have demonstrated, this prosaic notion is incorrect. The spherical nature of the Th(IV) cation allows it adopt to a broad range of coordination environments (6–15) allowing the formation of higher dimensional structures (at greater degree than that of uranium as example) which possess a range of chemical properties leading to pertinent applications in areas such as ion-exchange and catalysis.<sup>42–44</sup> Traditional studies of thorium compounds with high pressure or HP/HT techniques have typically focused on its binary oxide form in addition to hydrides and carbides.<sup>45–48</sup> The latter two examples were examined particularly in the context of potential superconducting materials. Many contemporary studies relating to thorium under HP/HT conditions have largely shifted focused towards utilising its flexible coordination chemistry to examine novel structure formation with a variety of inorganic ligand donors such as borate, tungstate and molybdate among others. Such works have added significant insight into the fundamental chemistry of thorium and its behaviour under HP/HT conditions. The following sections will briefly review recent success in the HP/HT studies of thorium compounds over the previous decade.

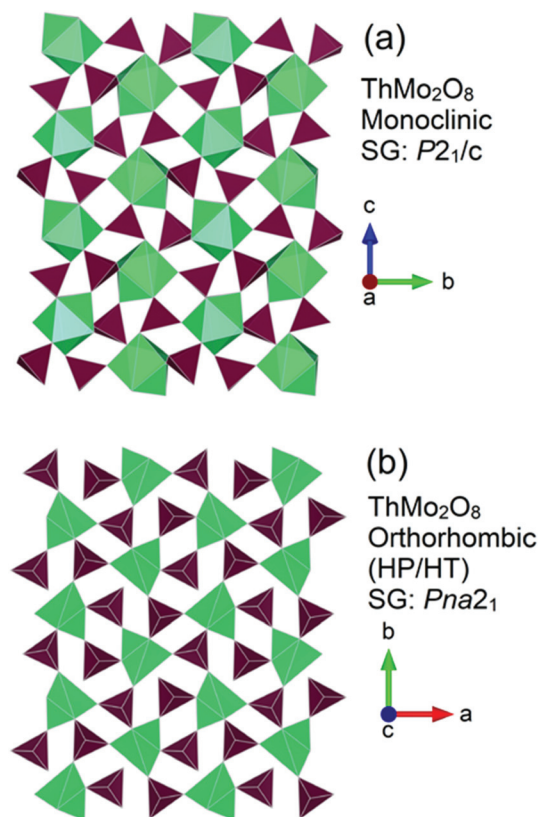
### Thorium molybdates

A pertinent example that illustrates the structural flexibility of thorium and how it can be manipulated through HP/HT conditions is exemplified by ThMo<sub>2</sub>O<sub>8</sub>.<sup>49</sup> At ambient pressure, using specific high temperature conditions ThMo<sub>2</sub>O<sub>8</sub> can adopt three different structural types, monoclinic (SG  $P2_1/c$ ), orthorhombic (SG  $Pbca$ ) and hexagonal (SG  $P\bar{3}$ ).<sup>49</sup> When exposed to conditions of over 3.5 GPa and 1000 °C an ortho-

rhombic variant (SG  $Pna2_1$ ) can be generated culminating in a reduction of molar volume by 20% compared to that of the ambient pressure monoclinic variant. Fig. 3 compares the general structures of the monoclinic and HP/HT orthorhombic variants.

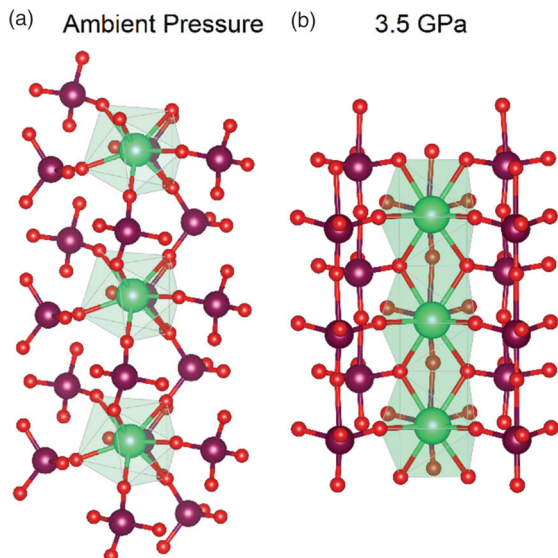
This significant reduction of volume in ThMo<sub>2</sub>O<sub>8</sub> is attributed to the extreme flexibility of the Th and Mo polyhedra. At ambient pressure, in the monoclinic form, like polyhedra are found approximately 6 Å apart in a linked geometry of Th-centres connected *via* MoO<sub>4</sub><sup>2-</sup> units in an approximately linear arrangement of Th and Mo cations (Fig. 4). Upon transformation to the HP/HT variant, the linear motif is lost. Instead, Th and Mo are now independently aligned and the distance between them has been reduced to 4 Å (Fig. 4).

Between the three ambient pressure polymorphs of ThMo<sub>2</sub>O<sub>8</sub> monoclinic, orthorhombic and hexagonal in addition to the HP/HT orthorhombic, the Th cation respectively adopts oxygen coordination environments of ThO<sub>8</sub>, ThO<sub>8</sub>, ThO<sub>6</sub>/ThO<sub>6</sub>/ThO<sub>9</sub> and ThO<sub>11</sub>. When contrasted against the broader family AB<sub>2</sub>O<sub>8</sub> structures (A = Zr, Hf and Th, B = Mo, W), the non-actinide equivalents have less structural diversity compared to ThMo<sub>2</sub>O<sub>8</sub>, only reported so far to occur with cubic structure types. This lack of diversity can be traced to the



**Fig. 3** General structure representations of the ThMo<sub>2</sub>O<sub>8</sub> polymorphs of (a) the ambient pressure high temperature synthesised monoclinic (SG  $P2_1/c$ ) and (b) the HP/HT orthorhombic (SG  $Pna2_1$ ). Note Th and Mo polyhedra are represented by green and maroon polygons respectively. Figure adapted from ref. 49.



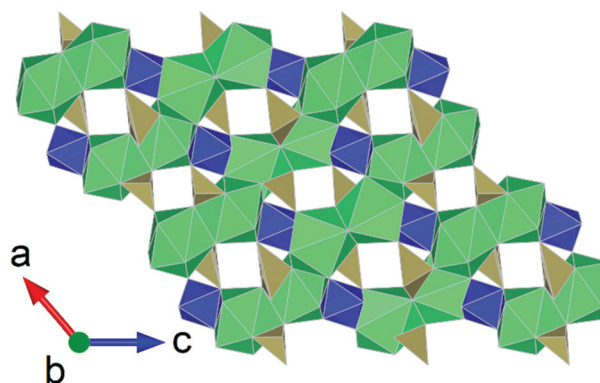


**Fig. 4** Comparison of the coordination motifs between the Th and Mo polyhedra at (a) ambient pressure synthesised monoclinic variant and (b) HP/HT orthorhombic variant of  $\text{ThMo}_2\text{O}_8$ . The figure illustrates the reduced distance between Th polyhedra between the ambient and higher-pressure variants. Note Th polyhedra, Mo and O atoms are represented by green polygons, maroon and red spheres respectively. Figure adapted from ref. 49.

largely restrictive coordination environments of the Zr and Hf cations which typically can only adopt coordination environments of 6–8, Th can take on coordination environments from 6–15.

### Thorium tellurates/tellurites

The oxyanions of tellurium, tellurate,  $\text{Te}(\text{iv})$ , and tellurite,  $\text{Te}(\text{vi})$ , have been utilised and explored extensively in inorganic synthesis due to their diverse chemical behaviour and that they can exist in a variety of geometries and coordination environments,  $\text{TeO}_x$  ( $x = 3\text{--}6$ ). The further presence of lone-pair second-order Jahn–Teller distorting electrons on the  $\text{Te}(\text{iv})$  cation leads to non-centrosymmetric structure (NSC) formation allowing desirable materials with second harmonic generation (SHG) and non-linear optical (NLO) properties to be obtained. Comparatively, tellurites can exist in a greater range of coordination geometries than tellurates where, as building units, their geometric flexibility leads to a variety structure being generated from 0D clusters, 1D chains, 2D layers to 3D frameworks with desirable functional properties. Utilisation of tellurate and tellurite chemistry with uranium and thorium has been somewhat focused on the former actinide with a variety of studies under conditions such as high temperature, hydrothermal, solvothermal, HP/HT and even many examples seen in minerals.<sup>20,50</sup> In contrast there are significantly less studies on thorium tellurate and tellurite compounds, no known related minerals and only one study under HP/HT conditions.<sup>51</sup> Nevertheless, the study by Xiao *et al.*<sup>51</sup> examined the formation of thorium–tellurium compounds through HP/HT



**Fig. 5** Structural representation of  $\text{Th}_2\text{Te}_3\text{O}_{11}$  highlighting coordination motif of different Te valences adapted from ref. 51. Note:  $\text{Th}(\text{iv})$ ,  $\text{Te}(\text{iv})$  and  $\text{Te}(\text{vi})$  polyhedra are presented as green, tan and blue polygons.

conditions of which  $\text{Th}_2\text{Te}_3\text{O}_{11}$  was uncovered, a mixed thorium tellurite–tellurate compound (Fig. 5). Unlike previous studies on thorium–tellurium compounds under ambient pressure conditions none have produced structures containing mixed Te cations. The occurrence of  $\text{Te}(\text{iv})$  and  $\text{Te}(\text{vi})$  in  $\text{Th}_2\text{Te}_3\text{O}_{11}$  further invokes the formation of mixed Te coordination's of  $\text{Te}(\text{iv})\text{O}_3$ ,  $\text{Te}(\text{iv})\text{O}_4$ , and  $\text{Te}(\text{vi})\text{O}_6$ . This multivalent and multicoordination possessing Te actinide compound was argued to be a consequence of the HP/HT conditions used, coupled with the flexibility of Te and Th coordination chemistry. The same study, further uncovered the first alkaline thorium tellurates of  $\text{Na}_4\text{Th}_2(\text{Te}(\text{vi})\text{O}_{15})$  and  $\text{K}_2\text{Th}(\text{Te}(\text{vi})\text{O}_4)_3$  in addition to the high pressure form of  $\text{Th}(\text{Te}(\text{iv})\text{O}_3)(\text{SO}_4)$ . The latter structure, compared to its lower pressure polymorph,<sup>52</sup> through HP/HT conditions involves an apparent total reconstructive phase transformation involving the Th coordination environment increasing from 8 to 9 and also dimensionality of the structure moving from 2D to 3D.

### Thorium arsenates/arsenites

Similar to tellurates and tellurites, actinide arsenic compounds have been investigated extensively due to their diversity in structure formation, a consequence of the  $\text{As}(\text{iii})$  and  $\text{As}(\text{v})$  cations being able to adopt multiple coordination environments and also being able to polymerise. The chemistry of arsenic is further of interest due to its toxicity and has accordingly garnered interest in the context of environmental safety.<sup>53</sup> Subsequently, studies have focused on examining the valence chemistry of arsenic compounds particularly with respect to forming mixed valent compounds and how actinides can coerce this. The first example of an actinide mixed valent arsenic compound was uncovered by Yu *et al.*<sup>54</sup> through applying HP/HT conditions to  $\text{Th}(\text{NO}_3)_4 \cdot 5\text{H}_2\text{O} \cdot \text{As}_2\text{O}_3 \cdot \text{CsNO}_3$  leading to the formation of  $\text{Th}(\text{As}(\text{iii})_4\text{As}(\text{v})_4\text{O}_{18})$ . Aside to this study some thorium arsenates/arsenites compounds had been reported from high temperature synthesis<sup>55,56</sup> but none with mixed arsenic valences. The structure of  $\text{Th}(\text{As}(\text{iii})_4\text{As}(\text{v})_4\text{O}_{18})$  consists of  $(\text{As}^{\text{III}}_4\text{As}^{\text{V}}_4\text{O}_{18})^{4-}$  layers that are interconnected *via*

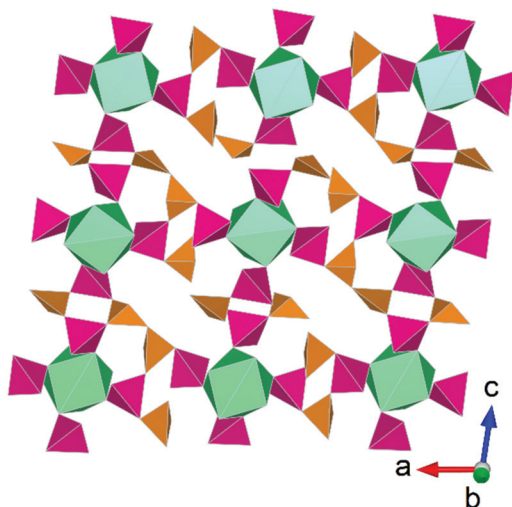


## Perspective

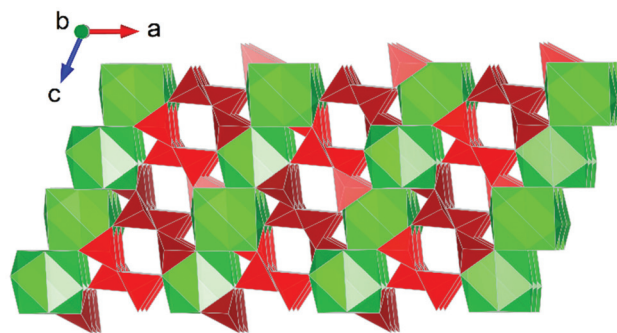
$\text{Th}^{4+}$  cations leading to a 3D framework structure. In ambient pressure studies of thorium arsenite/arsenate compounds,  $\text{AsO}_x$  units typically act as singular bridging units between  $\text{Th}^{4+}$  cations. Accordingly, the application of pressure through HP/HT synthesis is suggested to coerce polymerisation of  $\text{AsO}_x$  units and can be considered a tool in aiding this for synthesis design (Fig. 6).

## Thorium borates

Interest in actinide borate chemistry has a rich history, traditionally examined in the context of understanding interactions in glass-waste form matrices and immobilisation.<sup>17</sup> Contemporary studies have shifted their focus towards advanced functional material development. This shift is related to the flexibility in structure formation that borate units often provide in facilitating diverse actinide structure formation that typically features pores which can be used for ion-exchange applications,<sup>57</sup> notably involving thorium.<sup>44</sup> A number of HP/HT studies involving uranium borates have been conducted (discussed below), but only so far one study involving thorium borates under such conditions is reported.<sup>37</sup> The study by Hinteregger *et al.*<sup>37</sup> examined the formation of  $\text{AnB}_4\text{O}_8$  ( $\text{An} = \text{Th}, \text{U}$ ) under conditions of 5.5 GPa and 1100 °C for the thorium variant (uranium discussed below). The structure of  $\text{ThB}_4\text{O}_8$  was found to bare considerable resemblance to that of rare-Earth borates synthesised from HP/HT conditions, namely being built from tetrahedral  $\text{BO}_4$  groups as 2D layered structures.<sup>58,59</sup> However, in the case of  $\text{ThB}_4\text{O}_8$  it is found to be a 3D framework structure consisting of discrete channels (Fig. 7). As suggested by Hinteregger *et al.*<sup>37</sup> the occurrence of  $\text{Th}(\text{iv})$  and  $\text{U}(\text{iv})$  within the  $\text{AnB}_4\text{O}_8$  structure implies it should be possible for transuranium variants to be established, for



**Fig. 6** Structural representation of  $\text{Th}(\text{As}(\text{III})_4\text{As}(\text{V})_4\text{O}_{18})$  highlighting coordination motif of different As valences in the layer plane. Note that Th polyhedra (green) are coordinated firstly by  $\text{As}(\text{III})\text{O}_4$  (orange) forming discrete clusters that are linked by  $\text{As}(\text{V})\text{O}_3$  polyhedra (pink). Figure adapted from ref. 54.



**Fig. 7** Structural representation of the monoclinic channel structure of  $\text{AnB}_4\text{O}_8$  ( $\text{An} = \text{Th}, \text{U}$ ), (Th variant depicted). Note Th and B polyhedra are represented by green and red polygons respectively. Figure adapted from ref. 37.

instance containing  $\text{Pu}(\text{iv})$  (or as surrogate  $\text{Ce}(\text{iv})$ ). However, there are yet to be any reports of this.

## Uranium HP/HT studies

As the primary actinide element utilised within the nuclear fuel cycle, uranium has endured sustained interest in which publications pertaining to actinides are most associated with, and this holds further true for studies involving HP/HT conditions. Chemically, uranium can adopt a range of valence states from II–VI but of which the tetravalent ( $[\text{Rn}]5f^2$ ), pentavalent ( $[\text{Rn}]5f^1$ ) and hexavalent ( $[\text{Rn}]5f^0$ ) are most relevant for applications and solid-state materials.<sup>60</sup> The  $\text{U}(\text{v})$  state is traditionally considered a rarer intermediate between  $\text{U}(\text{iv})$  and  $\text{U}(\text{vi})$ , due to the unfavourable occupation of an unpaired single 5f electron in the 5f lowest unoccupied molecular orbital and thus will rapidly oxidise, reduce or disproportionate to retain a stable electronic configuration, particularly in aqueous state. However, contemporary investigations often by solid state and organometallic chemists had shed light on the ability to isolate  $\text{U}(\text{v})$  in solid compounds, through intricate synthesis methods including the use of pressure.<sup>20</sup> Compared to  $\text{U}(\text{iv})$ ,  $\text{U}(\text{vi})$  has an exceedingly higher solubility, considered greatly more reactive and accordingly is often studied in the context of environmental radiological contamination. Ubiquitous to their chemistry particularly in the solid state, the  $\text{U}(\text{v})$  and  $\text{U}(\text{vi})$  cations can be hosted within the oxo collinear uranyl group,  $\text{UO}_2^{2+}$ , but which is restricted for  $\text{U}(\text{iv})$ . The uranyl, a part of the general actinyl group, is pertinent to the chemistry of uranium and actinides and directly influences the structure dimensionality and properties of compounds. Its oxo nature typically leads to the formation of lower dimensional 2D layer or 1D chain structures as opposed to the spherical non-uranyl containing  $\text{U}(\text{iv})$  and  $\text{Th}(\text{iv})$  cations which adopt more higher dimensional structures. This subsequent preference in structure type formation based on uranium valence state provides then a basis for accurate and controlled synthesis design. Traditional studies involving HP/HT conditions of uranium



compounds, like thorium, largely focused on examining the pressure dependence of binary oxides like  $\text{UO}_2$  and also metallic using DACs in addition to some on mixed oxides as mentioned in the introduction previously.<sup>10,11</sup> However contemporary investigations on uranium materials under HP/HT conditions have expanded to broader classes of materials such as those of oxo-salt and mixed oxide compounds, particularly to utilise the relationship between uranium valence and structure type. Many of these HP/HT studies have become conducted in the context of alteration phases arising from SNF, advanced functional material development and fundamental explorative research of uranium chemistry. Accordingly, the next sections will briefly review and highlight successes of some of these contemporary investigations.

### Ternary uranium oxides

Interest in ternary uranium oxide compounds,  $\text{A}_x\text{U}_y\text{O}_z$ , (A = di- or trivalent cation) extends to the days of the Manhattan Project era with the early works of Zacharisen examining phases such as  $\text{SrUO}_4$  and  $\text{MgUO}_4$ .<sup>61,62</sup> Indeed, these simple ternary uranium oxides known as the monouranates of form the  $\text{AUO}_4$  have received the considerable interest in published literature and are still topical.<sup>4–6,60,63,64</sup> As previously mentioned, the  $\text{AUO}_4$  monouranates were some of the first actinide compounds to be examined using HP/HT methods in the case of the polymorphs of  $\text{NiUO}_4$  ( $\alpha$  and  $\beta$ ) examined by Young, Hoekstra and Marshall<sup>10,11</sup> in the 1960s. These studies suggested the inability of the  $\text{NiUO}_4$  polymorphs to form under ambient pressure high temperature conditions was a consequence of the size mismatch between the small Ni cation and that of the large uranium cation. Many of these traditional studies were often conducted with a sense of serendipity regarding acquiring structures and compounds from crudely designed synthesis experiments. Many contemporary investigations apply more systematic methods in synthesis design and analysis to better understand structure–property relationships. This was exemplified in a study re-examining the formation of structures from the Ni–U–O system, mono-,  $\text{NiUO}_4$ , di-,  $\text{NiU}_2\text{O}_6$ , and triuranates,  $\text{NiU}_3\text{O}_{10}$ , and particularly uncovering the need for HP/HT conditions for the synthesis of the monouranate variants.<sup>12</sup> The study using high-resolution measurements identified, consistent with the postulates of Brisi and Young, that HP/HT conditions are necessary to overcome the size mismatch between the Ni and U cations in the synthesis of the  $\text{NiUO}_4$  polymorphs. However further to this, structure formation can be achieved *via* excess U addition in place of pressure, which can bridge the size mismatch of the cations allowing the formation of the di and triuranates. This relationship between Ni–U–O structures, U/Ni ratio and pressure is graphically depicted in Fig. 8. W.r.t. the figure note the decreasing separation between Ni polyhedra with increasing pressure and reduced requirement for a higher H/Ni ratio to overcome this.

The structures of the  $\text{NiUO}_4$  polymorphs,  $\alpha$  and  $\beta$ , form respectively form rutile type structures of respectively 2D chain/layer and 1D chains structures which can be related to

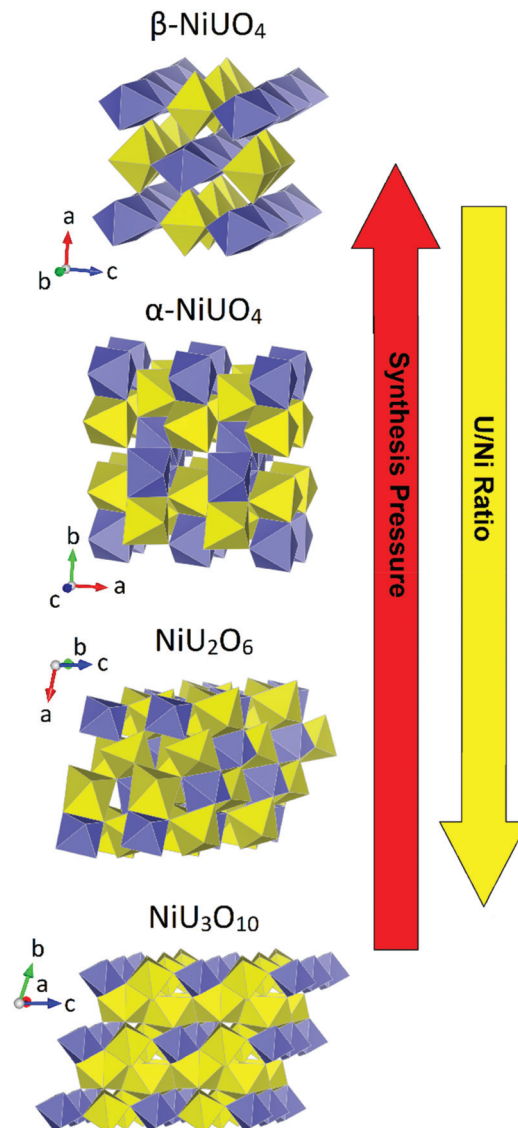


Fig. 8 Structural representations of mono-, di- and triuranates of the Ni–U–O system emphasising the relationship between the structure type and synthesis conditions. Note uranium and nickel polyhedra are given in yellow and purple respectively. Structures and conditions adapted from ref. 12.

$\alpha$ - $\text{PbO}_2$  and  $\text{TiO}_2$ .<sup>12</sup> A pertinent follow up study<sup>3</sup> found that by relating the rutile relationships present in the  $\text{NiUO}_4$  polymorphs with the broader family  $\text{AUO}_4$  monouranates which also possess rutile structures, subtle distortions and tilting in respective  $\text{UO}_6$  and  $\text{AO}_6$  polyhedra could be modulated through pressure, temperature or cation size. The study found that the  $\text{AUO}_4$  structures in space group  $Ibmm$ , A = Mn, Co, Mg and  $\beta$ -Ni which are related to  $\text{TiO}_2$ , can be potentially described in the higher symmetry space group  $Cmmm$  that  $\beta$ - $\text{CdUO}_4$  possesses, but the size mismatch between the U and A cations leads to structural distortions and symmetry lowering. However interestingly by applying temperature through *in situ* synchrotron powder diffraction heating measurements,

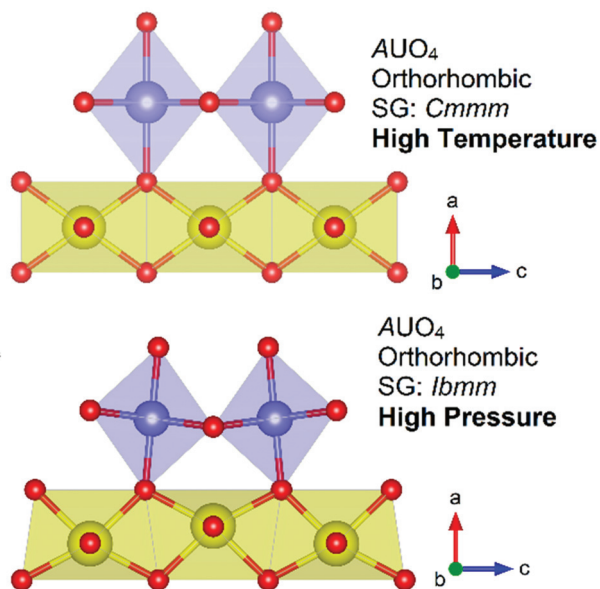


the higher symmetry non-distorted space group  $Cmmm$  structure could be obtained. Conversely by applying pressure through *ab initio* DFT +  $U$  simulations the lower symmetry distorted space group  $Ibmm$  structure could be obtained. Fig. 9 illustrates the difference in distortion of polyhedra and tilt between the distorted and non-distorted space group  $Ibmm$  and  $Cmmm$  structures respectively in addition to the relationship between pressure and temperature. The study gives a beautiful insight into the role temperature and pressure have upon modulating atomic position towards structure formation. The study<sup>3</sup> further saliently highlighted limitations in *ab initio* calculations using DFT +  $U$  for actinide metal oxides, often considered the gold standard for such computational work. The investigation found issues in applying the Hubbard  $U$  energy term which consistently resulted in the overestimation of the structural energy and consequently incorrect prediction of stability within the  $AUO_4$  compounds. The authors pointed towards the need to be better represent d and f states in calculations such as instead applying Wannier projections. Such insight is highly pertinent in the context of calculations relating to modern nuclear fuels which often use transition metals in their fuel blends.<sup>65</sup>

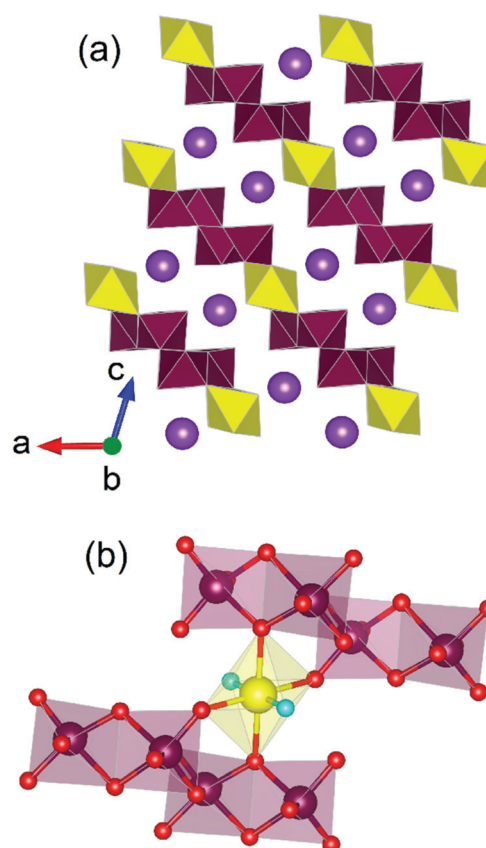
### Uranium molybdates

Uranium molybdates have been explored extensively for similar reasons to that of thorium counterparts, namely their contribution to desirable physical properties but also due to the pertinent occurrence of Mo in SNF.<sup>66</sup> However, unlike the spherical  $Th^{4+}$  cation, the oxo-uranyl group possessing  $UO_2^{2+}$

cation often leads to lower dimensional structure formation forming 1D chains or 2D layered structures over 3D framework. This was highlighted in so far the only study<sup>67</sup> of uranium molybdates synthesised under HP/HT conditions, which reported the compounds  $K_2[UO_2(Mo_2O_7)_2]$ ,  $K_2[(UO_2)_2(Mo(VI)_4Mo(IV)(OH)_2O_{16})]$ ,  $K_3[(UO_2)_6(OH)_2(MoO_4)_6(MoO_3OH)]$  and  $K_5[(UO_2)_{10}MoO_5O_{11}OH] \cdot H_2O$  from 900 °C and 4 GPa. Three of the four structures were found to adopt 3D framework structures with  $K_2[UO_2(Mo_2O_7)_2]$  being the exception adopting a 2D layered structure. Salient to this structure is the occurrence of oligomerised molybdate unit,  $(Mo_2O_7)^{2-}$ , which at time of writing has only observed in that structure and yet to be observed in potassium uranium molybdates resulting from more ambient pressure synthesis conditions (Fig. 10). It was highlighted by the study at time of publication that proportionally w.r.t. to other studies, HP/HT methods generally lead to a greater proportion of structures adopting 3D framework structures as opposed to high temperature and hydrothermal methods which use considerably lower pressures. The authors argued that both pressure and a high ratio of U/Mo units leads to the condensation of polyhedra resulting in the increased likelihood of oligomerisation of molybdate units, for instance



**Fig. 9** Structural representations of the  $AUO_4$  orthorhombic structures in space groups  $Cmmm$  and  $Ibmm$  where it has been shown<sup>3</sup> high temperatures favour the  $Cmmm$  structure non-distorted structure and high pressures favour the distorted  $Ibmm$  structure. Note the tilt in the  $AO_6$  polyhedra (purple) and distortion of  $UO_6$  polyhedra (yellow) w.r.t. oxygen atoms (red spheres).



**Fig. 10** (a) Structural layer motif of  $K_2[UO_2(Mo_2O_7)_2]$  highlighting the occurrence of (b) the  $(Mo_2O_7)^{2-}$  unit. Note maroon and yellow polyhedra respectively represent Mo and U polyhedra whereas purple, maroon, yellow, red and aqua spheres respectively represent K, Mo, U, O and uranyl-O atoms respectively. Figure adapted from ref. 67.





with the occurrence of  $\text{Mo}_2\text{O}_7^{2-}$  dimers and higher dimensionality of formed structure. Although, the study's sample size of known HP/HT derived uranium molybdates is relatively small, the postulate is well consistent with that which was similarly observed in thorium molybdates, ternary uranium oxides and in non-actinide molybdate based studies.<sup>68</sup> Consequently, the study points towards the ability of HP/HT methods to generate higher dimensional compounds and coerce oligomerisation of functional groups.

### Uranium tellurates/tellurites

Formation of uranium tellurates and tellurites was explored by Xiao *et al.*<sup>69</sup> under HP/HT conditions of 3.5 GPa and 1100 °C resulting in the structures  $\text{Na}_2[\text{UO}_2(\text{Te}(\text{vi})_2\text{O}_8)]$ ,  $\text{Na}_2[\text{UO}_2(\text{Te}(\text{vi})_2\text{O}_5)]$ ,  $\text{Na}[\text{UO}_2\text{Te}(\text{iv})_2\text{O}_5\text{OH}]$  and  $\text{Na}[\text{UO}_2\text{Te}(\text{iv})_6\text{O}_{13}\text{OH}]$ . Interestingly the structure of  $\text{Na}[\text{UO}_2\text{Te}(\text{iv})_6\text{O}_{13}\text{OH}]$  was found to be isotypical to the mineral cliffordite  $\text{UO}_2(\text{Te}_3\text{O}_7)$ ,<sup>50</sup> and crystallises in the centrosymmetric space group of  $Pa\bar{3}$ . Due to the oxo-bonding nature of the uranyl group, uranyl based compounds typically adopt NSC structures.<sup>70</sup> So far, there are no further studies of uranium tellurates and tellurites under HP/HT conditions.

### Uranium arsenates/arsenites

As previously described for thorium arsenates and arsenites, the uranium counterparts have been studied extensively using ambient pressure methods owing to the rich chemistry and diverse chemical behaviour that can be extracted from combining the chemistry of arsenic and uranium. Under high temperature ambient pressure and HP/HT conditions Yu *et al.*<sup>71</sup> examined the formation of caesium uranium arsenates with conditions of 900 °C and 3 GPa. The investigation led to four novel uranyl arsenates  $\text{Cs}_2[(\text{UO}_2)(\text{As}_2\text{O}_7)]$ ,  $\alpha\text{-Cs}[(\text{UO}_2)(\text{HAS}_2\text{O}_7)]$ ,  $\beta\text{-Cs}[(\text{UO}_2)(\text{HAS}_2\text{O}_7)]$  and  $\text{Cs}[(\text{UO}_2)(\text{HAS}_2\text{O}_7)]\cdot 0.17\text{H}_2\text{O}$  of which the latter three are the first examples of uranyl arsenates to be obtained under HP/HT conditions. However, there are yet to be any studies of uranyl arsenites resulting from HP/HT conditions. The study further supports the conclusions made on potassium uranyl molybdates under HP/HT<sup>67</sup> conditions previously discussed, that the application of pressure seems to assist in the transformation from 2D layered motif of  $\text{Cs}_2[(\text{UO}_2)(\text{As}_2\text{O}_7)]$  to that of framework structures of the high HP/HT synthesised phases. Accordingly, the investigation supports the narrative of HP/HT conditions being a driver towards increasing dimensionality of compounds.

### Uranium silicates

The elucidation of interactions and structure formation between uranium and silicon is a highly topical and long-examined area of actinide chemistry and mineralogy due to the strong association the elements have with SNF alteration phases and geochemistry.<sup>72,73</sup> It is known that uranium provides approximately 25% of the total energy generated within the Earth through its radioactive decay, and accordingly geochemists have investigated its speciation within Earth's

layers.<sup>24</sup> It is suspected that approximately 50% of terrestrial uranium is found within the lower mantle, where temperatures over 2000 °C and pressures above 50 GPa are experienced. Under these conditions, coffinite,  $\text{USiO}_4$ , is expected to decompose into  $\text{UO}_2$  and  $\text{SiO}_2$  and incorporation of the uranium oxide into region prevalent silicate perovskites including dave-maoite  $\text{CaSiO}_3$ ,<sup>74</sup> has been of strong topical interest. A pertinent investigation by Gréaux *et al.*<sup>75,76</sup> using HP/HT conditions provided by a laser heated DAC (LH-DAC) and using X-ray diffraction and spectroscopy, was able to show under conditions of up 54 GPa and 2400 °C, uranium incorporation into the  $\text{CaSiO}_3$  indeed occurs. The authors conclude consequently that Al rich  $\text{CaSiO}_3$  is the only candidate mineral able to host uranium providing key radiogenic heat (and also suspected thorium) in the Earth's mantle. Although the study is *in situ* based, it emphasises the importance of applying HP/HT conditions to actinide relevant minerals towards understanding geological behaviour of the Earth.

### Uranium borates

Of the actinide system studies that have been examined in the previous decade, uranium borates have arguably received the most attention and a large body of literature is available for them under a variety of conditions.<sup>17,77,78</sup> The ability for borate units to occur as seemingly an innumerable amount of fundamental building blocks through  $\text{BO}_3$  triangle and  $\text{BO}_4$  combinations that can also polymerise, gives a large breadth of investigation for chemists particularly using HP/HT conditions. Furthermore, as opposed to thorium borate studies, the chemistry of uranium allows even a greater diversity of structures to be obtained due to the greater accessibility of valence states of the latter actinide. In the previously discussed study by Hinteregger *et al.*<sup>37</sup> which examined the formation of  $\text{AnB}_4\text{O}_8$  (An = Th, U) structures, additional to the synthesis of thorium borate, they were also able to achieve synthesis of the uranium in its tetravalent state, despite beginning with hexavalent uranium nitrate as starting reagent. Compared to the thorium counterpart which used pressures and temperatures of 5.5 GPa and 1100 °C, a higher pressure of 10 GPa was employed. Although such a result is surprising considering the lack of reducing conditions used in their synthesis, a growing body evidence points towards the flexibility of uranium valence in facility structure formation based on size requirements. For example, in isostructural monouranates,  $\alpha\text{-NiUO}_4$ ,  $\text{CrUO}_4$  and  $\text{FeUO}_4$ , the adoption of the pentavalent uranium in the Cr and Fe members and hexavalent in Ni (synthesised under HP/HT conditions),<sup>12</sup> is argued to be based on structure and cation size ratio requirements, which is facilitated by the flexibility of the uranium valence to conform.<sup>3</sup> Similar is expected to occur in  $\text{UB}_4\text{O}_8$ , where the larger tetravalent uranium cation is needed over the smaller hexavalent of which the combination of extreme pressure and temperature through HP/HT synthesis facilitates its formation.

A number of studies in the early 2010's utilising supercritical water conditions<sup>79,80</sup> demonstrated the exotic and unique chemistry that can be obtained from elevated pressures and



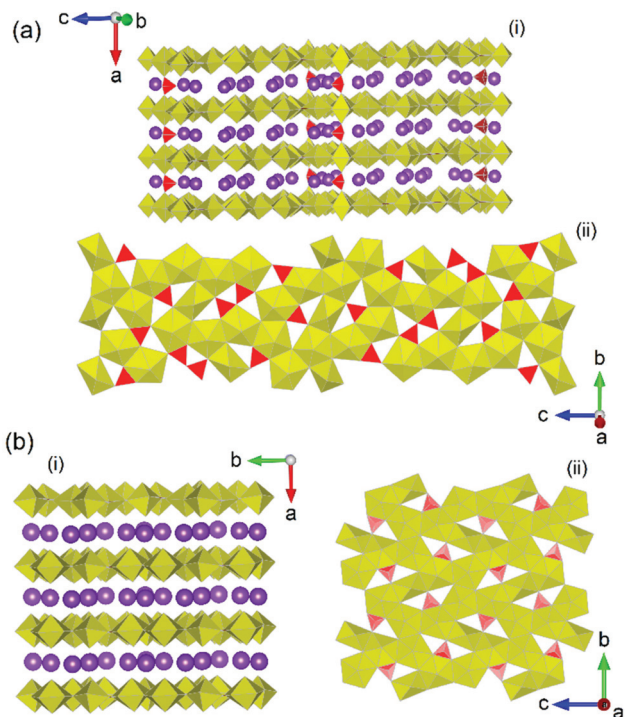
temperatures on uranyl borates. Inspired by these successes, researchers began examining uranyl borates under HP/HT conditions. A pertinent study by Wu *et al.*<sup>79</sup> examined the formation of uranyl borates at temperatures between 650 and 1200 °C and pressures between 0.2 and 2.5 GPa. The study unveiled three potassium uranyl borates which exhibited firstly a U/B ratio greater than 1 (rare for uranium borates) in addition to novel coordination geometries of uranyl centres that have not been encountered in ambient pressure studies. Two of the structures from the study,  $K_{12}[(UO_2)_{19}(UO_4)(B_2O_5)_2(BO_3)_6(BO_2OH)O_{10}] \cdot nH_2O$ , and  $K_4[(UO_2)_5(BO_3)_2O_4] \cdot H_2O$ , are presented in Fig. 11 which illustrates the starkly different layer configurations compared to somewhat similar structural configurations. The study eloquently demonstrated the novel coordination types and structural architectures which HP/HT conditions can provide for in uranyl borate synthesis. A later study by Hao *et al.*<sup>81</sup> added to this narrative, in using systematic methods to examine the formation of uranyl borates under high temperature, hydrothermal and HP/HT synthesis conditions. The study highlighted the variety of different uranyl borate layer topologies that can be obtained, in which the topology of the borate units is markedly contrasted, from different synthesis methods.

A topical sub-field of uranium borate chemistry is that of mixed anionic borate materials such as aluminoborates or

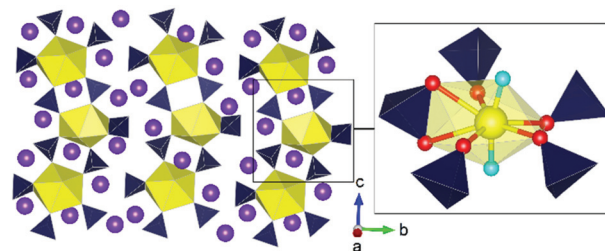
borophosphates.<sup>41,82</sup> Both types of uranium borates have been studied extensively in the context of nuclear waste form glasses but also because of the similar flexible bonding nature of phosphate and aluminate groups that are highly compatible with borate groups. This leads to an even greater degree of permutations and combinations that can be obtained when combined with uranium chemistry. The uranyl aluminoborate  $UO_2[B_3Al_4O_{11}(OH)]$  featuring a highly complex 3D structure was achieved by Wu *et al.*<sup>41</sup> using a cold seal vessel with conditions of 0.2 GPa and 650 °C. Pertinent to the structure is the occurrence of multi-oxo components of uranyl in addition to  $AlO_6$  and  $AlO_5$  groups that gives rise to its unique structure which deviates sharply from ambient pressure studies. Hinteregger *et al.*<sup>38</sup> examined the formation uranyl borophosphates under HP/HT conditions of 7.0 GPa and 700 °C. Similar to their previous study<sup>37</sup> on uranium borates, they were able to synthesis a tetravalent uranium borophosphate despite starting with hexavalent uranium nitrate. The structure,  $U_2[BO_4][PO_4]$ , interestingly was found to be isotypic to  $Th_2[BO_4][PO_4]$  which was obtained from high temperature ambient pressure conditions.<sup>83</sup> It is suspected the reason for the reduction of the uranium during synthesis is again related to cation size requirement in structure formation, of which the flexibility of uranium redox chemistry can accommodate this through HP/HT conditions.

#### Uranium sulfates

There is so far only one such study of uranium sulfates under HP/HT conditions.<sup>84</sup> As is often the result of fundamental investigations involving extreme conditions, the study describes the serendipitous synthesis of a HP/HT uranium sulfate phase,  $Na_4[(UO_2)(SO_4)_3]$ , from a synthesis at 3.5 GPa and 1000 °C that was intended to examine the formation of actinide oxo-chalcogenates. Pertinently, the structure possesses the uranyl group, which is typically linear with only small angular deviations,<sup>85</sup> with a pronounced bend away from 180° to 165.6(12)° (Fig. 12). Significant efforts have been made in general uranium and actinide chemistry to understand the behaviour of the uranyl/actinyl group regarding reactivity such as in cation–cation interactions (CCIs) or particularly examining limitations in bending and its deformation. To probe the



**Fig. 11** Structural representations of (a) orthorhombic  $K_{12}[(UO_2)_{19}(UO_4)(B_2O_5)_2(BO_3)_6(BO_2OH)O_{10}] \cdot nH_2O$  in (i) general and (ii) layer format, and (b) monoclinic  $K_4[(UO_2)_5(BO_3)_2O_4] \cdot H_2O$  in (i) general and (ii) layer formats. Yellow and red polyhedra represent U and B polyhedra respectively whereas purple spheres represent K atoms. Figures adapted from ref. 79.



**Fig. 12** Structure of  $Na_4[(UO_2)(SO_4)_3]$  synthesised from HP/HT conditions illustrating bending of the uranyl group in the inset. Note Yellow and dark polyhedra represent U and S polyhedra respectively whereas purple, yellow, red and aqua blue spheres represent K, U, O and uranyl O atoms respectively. Figure adapted from ref. 84.



latter phenomena, chemists have often used scaffolding methods often with organometallic ligands to induce bending or deformation that is steric or electronic in origin.<sup>86</sup> At present, the maximum uranyl bend reported is approximately 161°. <sup>87,88</sup> In the case of Na<sub>4</sub>[(UO<sub>2</sub>)(SO<sub>4</sub>)<sub>3</sub>], the authors argue that bending originates from the physical pressure applied to the structure rather than an electronic or steric effect. The latter effects arise from large steric groups pushing the uranyl, of which sulphate groups do not possess such strength. Rather, the sulphate groups are strong equatorial donors, which can weaken the uranyl moiety, leading to its bending when placed under HP/HT conditions during structure formation and subsequent quenching. The authors used *ab initio* calculations to support these conclusions. The study is significant, as it offers a new pathway towards understanding the fundamental chemistry of the uranyl group, of which was only previously examined using organic and organometallic ligands.

## Transuranics

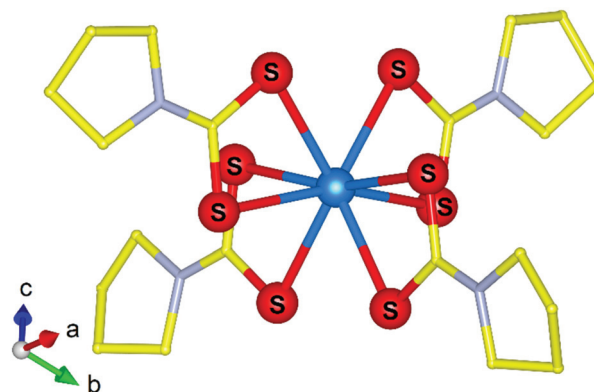
Despite the significant advances made in the chemistry of uranium and thorium under HP/HT conditions, little progress has been made for transuranium element containing compounds. A consequence of the heightened radiological danger associated with the elements coupled with the unavailability of HP/HT apparatuses and laboratories that are licensed and can accommodate them. Despite this, the last decade has unveiled considerable new insight into the chemistry of transuranium compounds, divulging significant chemical insight particularly under conditions of elevated pressure *via* pertinent studies involving DACs. Accordingly, some of these will be mentioned briefly considering their salience to actinide chemistry.

Plutonium metals have been examined extensively at higher pressures (and also temperatures) due to the complexity in their electronic structure which leads to a plethora of allotropes.<sup>89</sup> In the last decade several papers have examined the associated pressure and temperature induced transitions. A long-standing question of plutonium chemistry has been the observation of anomalous large softening of the bulk modulus in the metallic form as temperature increases irrespective of the sign of the coefficient of thermal expansion, which is not encountered to such an extent as other metals. However it was recently shown by Harrison<sup>90</sup> that the softening of the bulk modulus can be attributed to the compressibility of plutonium's electronic configuration. Such phenomena arises from the unique ability for plutonium to adopt large number of electronic configurations that are non-degenerate.<sup>91</sup> This result obtained through statistical thermodynamic modelling, underscores the exotic chemistry of plutonium and that despite 82 years since its discovery, key questions remain behind its HP/HT behaviour even its metallic form.

Transplutonium elements, unlike their lighter actinide sisters, possess unique localisation of 5f electrons at ambient pressures but delocalise and partake in bonding at higher

pressures. This results in a rich array of crystallographic structures that form under elevated pressure which exhibit intriguing structural and magnetic properties.<sup>92,93</sup> This is well exemplified by curium which has been examined extensively under high pressure conditions over the last decades and several allotropes idealised.<sup>94</sup> Yet, still the story is not over with recent discoveries occurring under elevated pressure conditions such as the predicted occurrence of quasiparticle multiplets at high pressure.<sup>95</sup> Further pioneering studies into californium by Heathman *et al.*<sup>96</sup> found in 2013 that californium to be first actinide to exhibit more than one metallic valence (from divalent to trivalent) near ambient conditions from *in situ* pressure studies. This phenomenon is due to the strong ability for californium ability to delocalise its f electrons under pressure allowing them to adopt different valence states.

It has often been thought close parallels can be made between the electronic characteristics of heavy 4f lanthanide elements with that of the 5f transplutonium and by extension bonding characteristics. However, over the last decade this perception has been rapidly reduced through pioneering transplutonium studies by the group of Albrecht-Schönzart utilising optical spectroscopy measurements under pressure using DACs. A study by Sperling *et al.*<sup>97</sup> examined the compression of curium pyrrolidinedithiocarbamate (pydtc) along with its neodymium analogue using absorption spectroscopy under pressures up to 11 GPa. They showed that the f–f transitions in [Cm(pydtc)<sub>4</sub>]<sup>−</sup> (Fig. 13) with its Cm–S bonds are more considerable than that of the Nd analogue and also Cm mellitate with its Cm–O bonds. The study highlighted the complexity of curium bonding, particularly covalent interactions at high pressure and the unexpected further impact it has on the ligands. Similar results were also reported by Sperling *et al.*<sup>98–100</sup> on related americium in addition to similar berkelium and californium compounds by like methods highlighting both the sharp departure in bonding behaviour compared to non-actinide variants. Paucities of information still remain for the transuranics under pressure both from the understand-



**Fig. 13** Molecular structure representation of the [Cm(pydtc)<sub>4</sub>]<sup>−</sup> anion that was examined by Sperling *et al.*<sup>97</sup> using absorption spectroscopy *via* DACs. Note Cm, S, N and C atoms are represented by blue, red, silver and yellow spheres respectively. Figure adapted from ref. 97.



ing of the fundamental electronic states to their bonding with compounds. Nevertheless, significant advancements have been made in both areas which have enabled deeper understanding to their core chemical properties and bonding behaviours.

## Perspectives, challenges, and future directions

From this overview, over the last decade *ex situ* HP/HT studies of actinide compounds have allowed significant and novel understanding of their physical–chemical properties to be obtained. For instance, it is now more comfortably understood that when considering the results of potassium uranyl molybdates,<sup>67</sup> arsenates/arsenites<sup>71</sup> and thorium tellurates/tellurites,<sup>51</sup> HP/HT conditions, particularly pressure, enable strong control to be obtained over the dimensionality of structure formation. Specifically, higher pressure conditions favour high dimensional structures. Indeed, this seems to extend further with the coerced oligomerisation of building units with the application of pressure and temperature. Another interesting observation seems to be the unexpected flexibility of actinide valence states under pressure, emphasised in uranium studies<sup>12,37,38</sup> where despite beginning with oxidised uranium either reduced forms are found after HP/HT conditions are applied or structure formation seems to diverge from what is expected in ambient pressure studies regarding uranium valence.<sup>63</sup> Generally, it can be observed that HP/HT conditions provide an open playground in which actinide compounds are able to exhibit flexible and exotic structure formation, through the adoption of multiple polymorphs at elevated pressures, exhibit distinctive bonding and polyhedra distortions and sometimes unexpected valence behaviour.

It is also quite evident from the discussion thus far that several thorium, uranium and particularly transuranium chemical systems are underexplored. For instance, thorium compounds of phosphate, tungstate, iodate and selenate/selenite have received broad interest under variable temperature and ambient to low pressure conditions, but there are no reports of these from HP/HT conditions.<sup>101–103</sup> Similarly for uranium, compounds of iodate, phosphate, tungstate and selenate/selenite among others have not been reported under HT/HP conditions, despite considerable general interest.<sup>104–107</sup> In the case of transuranium compounds, there are yet to be any reports of compounds from HP/HT conditions. Although, the works of Albrecht-Schönzart and co-workers have hinted at the unique bonding characteristics of transuranium compounds under pressure, unfortunately there appears to be no laboratories operating that are equipped for HP/HT experimentation with transuranium elements at this time. This absence is more related to the lack and associated cost of HP/HT infrastructure, rather than transuranium laboratories. Nevertheless, the innovative use of DAC devices in transuranium laboratories has offered a somewhat alternative to this, and it is hoped then that LH-DAC can provide a relatively inexpensive substitute to HP/

HT devices and find further application. There is a further general absence of studies pertaining to pressures above 10 GPa for actinide compounds. Although piston cylinder/multi-anvil devices such as that describe previously (Fig. 1), can reach such pressures, it is generally an expensive exercise particularly w.r.t. time. However, as demonstrated by geochemical studies such as the speciation of uranium in the Earth's mantle,<sup>74</sup> there is an inherent interest to examine compounds and materials arising under such conditions in quantities larger than what a DAC can produce. Accordingly, despite the advancements in DAC measurements there is an on-going need for devices that can produce larger size samples from HP/HT conditions such as that described and presented in Fig. 1.

It has been previously established that HP/HT conditions are an alternative means to produce good quality single crystals, particularly from materials that have high melting points.<sup>108,109</sup> Indeed, actinide materials, particularly those of oxide or silicate such as UO<sub>2</sub> and coffinite are notoriously difficult to generate single crystals of.<sup>73</sup> The synthesis of coffinite even as a powder is a difficult task,<sup>73</sup> often requiring intricate synthesis methods involving elevated pressure *via* hydrothermal conditions to synthesis it,<sup>72,73,110,111</sup> indeed there are no further reports of single crystal growth. In the case of UO<sub>2</sub>, methods such as arc-melting, floating zone techniques and vapor deposition are required, which many actinide laboratories do not possess.<sup>112</sup> Accordingly, it is argued that combined with appropriate fluxes, HP/HT conditions can be used as an alternative means to produce often sort after actinide single crystals such as UO<sub>2</sub> and potentially more difficult to obtain, coffinite and related silicate minerals.

A topical area of actinide chemistry recently has been examining the formation and properties of actinide metal organic framework (MOF) compounds. MOFs have received long standing interest due to their notable applications in hydrogen and carbon storage, electrocatalysis, semiconductors, gas separation among others. Placing MOFs under pressure has often been conducted to understand load bearing, deformation, and pressure-induced amorphisation of MOFs but it has also been found they can exhibit extreme compressibility.<sup>113</sup> Studies relating to actinide MOFs have largely focused on topics of sequestration and capture of radionuclides in addition to fundamental synthesis studies of MOFs using actinides as building blocks. It is postulated HP/HT conditions would be then fertile ground for MOFs synthesis in utilising the unique chemical properties of actinide bonding under pressure to unlock novel structures and properties, as has been achieved in non-actinide HP/HT studies of MOFs.<sup>114,115</sup>

## Concluding remarks

Overall, HP/HT studies of uranium and thorium compounds have uncovered unique and unexpected structure formation and bonding motifs. Whereas those of transuranium compounds under elevated pressure highlight the starkly different



chemical properties of heavy actinide elements compared to relative lanthanides and lighter actinides. From these studies, a good basis of chemical knowledge of actinides under HP/HT conditions has been established. Nevertheless, it is important that this fundamental knowledge finds suitable use and application in areas of geoscience, nuclear waste form/NFC materials and advanced functional development. The latter subject has particularly gained considerable momentum by Chinese research groups, demonstrating profound properties in a variety of materials from ion-conductors, SHG materials to radiation detectors.<sup>116–118</sup> However importantly, the developments of the fundamental understanding behind many HP/HT study conclusions began only approximately 10 years ago when a firm basis of actinide chemistry under ambient pressure and medium to high temperature conditions was already established. Accordingly, it can be viewed that HP/HT science of actinides is still a few years behind that of ambient pressure studies. Nevertheless, the renewed interest in actinide chemistry particularly in the context of reinvigoration of nuclear energy and overhauling nuclear waste disposal is expected to drive research back into HP/HT related studies, towards supporting science related to the NFC and fundamental discovery and materials development.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

GLM is grateful to funding and support from the German Federal Ministry of Education and Research (BMBF), Project No. 02NUK060. EVA and GLM are grateful to the funding and support from the German Science Foundation, DFG, Project No. AL1527/3-1.

## References

- 1 P. A. M. Dirac, *Proc. R. Soc. London, Ser. A*, 1926, **112**, 661–677.
- 2 S. Knecht, H. J. A. Jensen and T. Saue, *Nat. Chem.*, 2019, **11**, 40–44.
- 3 G. L. Murphy, Z. Zhang, R. Tesch, P. M. Kowalski, M. Avdeev, E. Y. Kuo, D. J. Gregg, P. Kegler, E. V. Alekseev and B. J. Kennedy, *Inorg. Chem.*, 2021, **60**, 2246–2260.
- 4 G. L. Murphy, C. H. Wang, Z. M. Zhang, P. M. Kowalski, G. Beridze, M. Avdeev, O. Muransky, H. E. A. Brand, Q. F. Gu and B. J. Kennedy, *Inorg. Chem.*, 2019, **58**, 6143–6154.
- 5 G. L. Murphy, C.-H. Wang, G. Beridze, Z. Zhang, J. A. Kimpton, M. Avdeev, P. M. Kowalski and B. J. Kennedy, *Inorg. Chem.*, 2018, **57**, 5948–5958.
- 6 G. L. Murphy, B. J. Kennedy, J. A. Kimpton, Q. Gu, B. Johannessen, G. Beridze, P. M. Kowalski, D. Bosbach, M. Avdeev and Z. Zhang, *Inorg. Chem.*, 2016, **55**, 9329–9334.
- 7 P. M. Kowalski, G. Beridze, Y. Ji and Y. Li, *MRS Adv.*, 2017, **2**, 491–497.
- 8 T. Connor, O. Cheong, T. Bornhake, R. Tesch, M. Sun, Z. He, A. Bukayemsky, V. Vinograd, S. C. Finkeldei and P. M. Kowalski, *Front. Chem.*, 2021, 940.
- 9 G. Beridze and P. M. Kowalski, *J. Phys. Chem. A*, 2014, **118**, 11797–11810.
- 10 A. Young, *Science*, 1966, **153**, 1380–1381.
- 11 H. R. Hoekstra and R. H. Marshall, *Some uranium-transition element double oxides*, ACS Publications, 1967.
- 12 G. L. Murphy, P. Kegler, Y. Zhang, Z. Zhang, E. V. Alekseev, M. D. de Jonge and B. J. Kennedy, *Inorg. Chem.*, 2018, **57**, 13847–13858.
- 13 C. Weir, E. Lippincott, A. Van Valkenburg and E. Bunting, *J. Res. Natl. Bur. Stand., Sect. A*, 1959, **63**, 55.
- 14 M. Idiri, T. Le Bihan, S. Heathman and J. Rebizant, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2004, **70**, 014113.
- 15 M. McElfresh, J. Thompson, J. Willis, M. Maple, T. Kohara and M. Torikachvili, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1987, **35**, 43.
- 16 N. Tateiwa, T. Kobayashi, K. Hanazono, K. Amaya, Y. Haga, R. Settai and Y. Onuki, *J. Phys.: Condens. Matter*, 2001, **13**, L17.
- 17 S. A. Wang, E. V. Alekseev, W. Depmeier and T. E. Albrecht-Schmitt, *Chem. Commun.*, 2011, **47**, 10874–10885.
- 18 N. N. Krot and M. S. Grigoriev, *Usp. Khim.*, 2004, **73**, 94–106.
- 19 B. I. Kharisov and M. A. Mendez-Rojas, *Usp. Khim.*, 2001, **70**, 974–995.
- 20 C. R. Graves and J. L. Kiplinger, *Chem. Commun.*, 2009, 3831–3853, DOI: [10.1039/b902969a](https://doi.org/10.1039/b902969a).
- 21 J. Drozdzyński, *Coord. Chem. Rev.*, 2005, **249**, 2351–2373.
- 22 S. Taylor, *Geochim. Cosmochim. Acta*, 1964, **28**, 1273–1285.
- 23 K. Lodders and B. Fegley, *The planetary scientist's companion*, Oxford University Press on Demand, 1998.
- 24 E. R. Oxburgh and R. K. O'Nions, *Science*, 1987, **237**, 1583–1588.
- 25 R. B. Rebak, *Accident-Tolerant Materials for Light Water Reactor Fuels*, Elsevier, 2020.
- 26 M. Ruzickova, T. Schulenberg, D. C. Visser, R. Novotny, A. Kiss, C. Maraczy and A. Toivonen, *Prog. Nucl. Energy*, 2014, **77**, 381–389.
- 27 T. Abe and K. Asakura, *Comprehensive Nuclear Materials*, ed. R. J. M. Konings, Elsevier, Oxford, 2012, pp. 393–422. DOI: [10.1016/B978-0-08-056033-5.00036-7](https://doi.org/10.1016/B978-0-08-056033-5.00036-7).
- 28 L. Van Brutzel, A. Chartier, Mechanical behaviour of UO<sub>2</sub> under irradiation: a molecular dynamics study, *MMSNF (The Materials Modelling and Simulation for Nuclear Fuels)*, 2017.
- 29 C. Lee, J. Lee, S. Park, S. Kwon, W.-J. Cho and G. Y. Kim, *Tunn. Undergr. Space Technol.*, 2020, **103**, 103452.
- 30 A. J. Beswick, F. G. Gibb and K. P. Travis, *Proc. Inst. Civ. Eng.: Energy*, 2014, **167**, 47–66.



- 31 V. Chazel, P. Gerasimo, V. Debouis, P. Laroche and F. Paquet, *Radiat. Prot. Dosim.*, 2003, **105**, 163–166.
- 32 D. E. Crean, M. C. Stennett, F. R. Livens, D. Grolimund, C. N. Borca and N. C. Hyatt, *Environ. Sci.: Processes Impacts*, 2020, **22**, 1577–1585.
- 33 G. N. Eby, N. Charnley, D. Pirrie, R. Hermes, J. Smoliga and G. Rollinson, *Am. Mineral.*, 2015, **100**, 427–441.
- 34 N. Eby, R. Hermes, N. Charnley and J. A. Smoliga, *Geol. Today*, 2010, **26**, 180–185.
- 35 D. Walker, M. Carpenter and C. Hitch, *Am. Mineral.*, 1990, **75**, 1020–1028.
- 36 J. R. Holloway and B. J. Wood, *Simulating the Earth: experimental geochemistry*, Springer Science & Business Media, 2012.
- 37 E. Hinteregger, T. S. Hofer, G. Heymann, L. Perfler, F. Kraus and H. Huppertz, *Chem. – Eur. J.*, 2013, **19**, 15985–15992.
- 38 E. Hinteregger, K. Wurst, L. Perfler, F. Kraus and H. Huppertz, *Eur. J. Inorg. Chem.*, 2013, 5247–5252.
- 39 H. Huppertz, *Chem. Commun.*, 2011, **47**, 131–140.
- 40 H. Huppertz, *Z. für Krist. – Cryst. Mater.*, 2004, **219**, 330–338.
- 41 S. Wu, O. Beermann, S. Wang, A. Holzheid, W. Depmeier, T. Malcherek, G. Modolo, E. V. Alekseev and T. E. Albrecht-Schmitt, *Chem. – Eur. J.*, 2012, **18**, 4166–4169.
- 42 M. E. Garner, B. F. Parker, S. Hohloch, R. G. Bergman and J. Arnold, *J. Am. Chem. Soc.*, 2017, **139**, 12935–12938.
- 43 J. L. R. Costa, G. S. Marchetti and M. do Carmo Rangel, *Catal. Today*, 2002, **77**, 205–213.
- 44 S. A. Wang, E. V. Alekseev, D. W. Juan, W. H. Casey, B. L. Phillips, W. Depmeier and T. E. Albrecht-Schmitt, *Angew. Chem., Int. Ed.*, 2010, **49**, 1057–1060.
- 45 C. Satterthwaite and I. Toepke, *Phys. Rev. Lett.*, 1970, **25**, 741.
- 46 M. Krupka, A. Giorgi, N. Krikorian and E. Szklarz, *J. Less-Common Met.*, 1969, **19**, 113–119.
- 47 J.-P. Dancausse, E. Gering, S. Heathman and U. Benedict, *High Pressure Res.*, 1990, **2**, 381–389.
- 48 A. Jayaraman, G. Kourouklis and L. Van Uitert, *Pramana*, 1988, **30**, 225–231.
- 49 B. Xiao, P. Kegler, T. M. Gesing, L. Robben, A. Blanca-Romero, P. M. Kowalski, Y. Li, V. Klepov, D. Bosbach and E. V. Alekseev, *Chem. – Eur. J.*, 2016, **22**, 946–958.
- 50 R. V. Gaines, *Am. Mineral.*, 1969, **54**, 697–701.
- 51 B. Xiao, P. Kegler, D. Bosbach and E. V. Alekseev, *Inorg. Chem.*, 2017, **56**, 2926–2935.
- 52 J. Lin, J. N. Cross, J. Diwu, N. A. Meredith and T. E. Albrecht-Schmitt, *Inorg. Chem.*, 2013, **52**, 4277–4281.
- 53 P. Bhattacharya, A. H. Welch, K. G. Stollenwerk, M. J. McLaughlin, J. Bundschuh and G. Panaullah, *Journal*, 2007, **379**, 109–120.
- 54 N. Yu, V. V. Klepov, P. Kegler, D. Bosbach, T. E. Albrecht-Schmitt and E. V. Alekseev, *Inorg. Chem.*, 2014, **53**, 8194–8196.
- 55 Y. Hao, E. V. Alekseev, V. V. Klepov and N. Yu, *Eur. J. Inorg. Chem.*, 2020, 3187–3193.
- 56 N. Yu, V. V. Klepov, H. Schlenz, D. Bosbach, P. M. Kowalski, Y. Li and E. V. Alekseev, *Cryst. Growth Des.*, 2017, **17**, 1339–1346.
- 57 S. Wu, S. Wang, A. Simonetti, F. Chen and T. E. Albrecht-Schmitt, *Radiochim. Acta*, 2011, **99**, 573–579.
- 58 G. Heymann, T. Soltner and H. Huppertz, *Solid State Sci.*, 2006, **8**, 821–829.
- 59 A. Haberer, G. Heymann and H. Huppertz, *Z. Naturforsch. B*, 2007, **62**, 759–764.
- 60 G. L. Murphy, Z. Zhang and B. J. Kennedy, *Complex Oxides: An Introduction*, WORLD SCIENTIFIC, Singapore, 2019, pp. 103–130.
- 61 W. H. Zachariasen, *Acta Crystallogr.*, 1954, **7**, 788–791.
- 62 W. H. Zachariasen, *Acta Crystallogr.*, 1948, **1**, 281–285.
- 63 X. F. Guo, E. Tiferet, L. Qi, J. M. Solomon, A. Lanzirrotti, M. Newville, M. H. Engelhard, R. K. Kukkadapu, D. Wu, E. S. Ilton, M. Asta, S. R. Sutton, H. W. Xu and A. Navrotsky, *Dalton Trans.*, 2016, **45**, 4622–4632.
- 64 G. Murphy, B. J. Kennedy, B. Johannessen, J. A. Kimpton, M. Avdeev, C. S. Griffith, G. J. Thorogood and Z. M. Zhang, *J. Solid State Chem.*, 2016, **237**, 86–92.
- 65 P. Kegler, M. Klinkenberg, A. Bukaemskiy, G. L. Murphy, G. Deissmann, F. Brandt and D. Bosbach, *Materials*, 2021, **14**, 6160.
- 66 L. Johnson, C. Ferry, C. Poinssot and P. Lovera, *J. Nucl. Mater.*, 2005, **346**, 56–65.
- 67 G. L. Murphy, P. Kegler, M. Klinkenberg, S. Wang and E. V. Alekseev, *Dalton Trans.*, 2020, **49**, 15843–15853.
- 68 A. Jayaraman, S. Sharma, S. Wang, S. Shieh, L. Ming and S. W. Cheong, *J. Raman Spectrosc.*, 1996, **27**, 485–490.
- 69 B. Xiao, P. Kegler, D. Bosbach and E. V. Alekseev, *Inorg. Chem.*, 2016, **55**, 4626–4635.
- 70 G. L. Murphy, E. M. Langer, O. Walter, Y. Wang, S. Wang and E. V. Alekseev, *Inorg. Chem.*, 2020, **59**, 7204–7215.
- 71 N. Yu, P. Kegler, V. V. Klepov, J. Dellen, H. Schlenz, E. M. Langer, D. Bosbach and E. V. Alekseev, *Dalton Trans.*, 2015, **44**, 20735–20744.
- 72 X. Guo, S. Szenknect, A. Mesbah, N. Clavier, C. Poinssot, S. V. Ushakov, H. Curtius, D. Bosbach, R. C. Ewing and P. C. Burns, *Proc. Natl. Acad. Sci. U. S. A.*, 2015, **112**, 6551–6555.
- 73 A. Mesbah, S. Szenknect, N. Clavier, J. Lozano-Rodriguez, C. Poinssot, C. Den Auwer, R. C. Ewing and N. Dacheux, *Inorg. Chem.*, 2015, **54**, 6687–6696.
- 74 O. Tschauner, S. Huang, S. Yang, M. Humayun, W. Liu, S. N. Gilbert Corder, H. A. Bechtel, J. Tischler and G. R. Rossman, *Science*, 2021, **374**, 891–894.
- 75 S. Greaux, L. Gautron, D. Andrault, N. Bolfan-Casanova, N. Guignot and M. A. Bouhifd, *Phys. Earth Planet. Inter.*, 2009, **174**, 254–263.
- 76 S. Gréaux, F. Farges, L. Gautron, N. Trcera, A.-M. Flank and P. Lagarde, *Am. Mineral.*, 2012, **97**, 100–109.
- 77 M. A. Silver and T. E. Albrecht-Schmitt, *Coord. Chem. Rev.*, 2016, **323**, 36–51.



- 78 G. Yuan and D. Xue, *Acta Crystallogr., Sect. B: Struct. Sci.*, 2007, **63**, 353–362.
- 79 S. Wu, S. Wang, M. Polinski, O. Beermann, P. Kegler, T. Malcherek, A. Holzheid, W. Depmeier, D. Bosbach, T. E. Albrecht-Schmitt and E. V. Alekseev, *Inorg. Chem.*, 2013, **52**, 5110–5118.
- 80 J. T. Stritzinger, E. V. Alekseev, M. J. Polinski, J. N. Cross, T. M. Eaton and T. E. Albrecht-Schmitt, *Inorg. Chem.*, 2014, **53**, 5294–5299.
- 81 Y. Hao, P. Kegler, T. E. Albrecht-Schmitt, S. Wang, Q. Dong and E. V. Alekseev, *Eur. J. Inorg. Chem.*, 2020, 407–416.
- 82 Y. Hao, G. L. Murphy, D. Bosbach, G. Modolo, T. E. Albrecht-Schmitt and E. V. Alekseev, *Inorg. Chem.*, 2017, **56**, 9311–9320.
- 83 C. Lipp and P. C. Burns, *Can. Mineral.*, 2011, **49**, 1211–1220.
- 84 E. M. Langer, P. Kegler, P. M. Kowalski, S. Wang and E. V. Alekseev, *Inorg. Chem.*, 2021, **60**, 8419–8422.
- 85 R. J. Baker, *Chem. – Eur. J.*, 2012, **18**, 16258–16271.
- 86 T. W. Hayton, *Dalton Trans.*, 2018, **47**, 1003–1009.
- 87 E. A. Pedrick, J. W. Schultz, G. Wu, L. M. Mirica and T. W. Hayton, *Inorg. Chem.*, 2016, **55**, 5693–5701.
- 88 S. Schöne, T. Radoske, J. März, T. Stumpf, M. Patzschke and A. Ikeda-Ohno, *Chem. – Eur. J.*, 2017, **23**, 13574–13578.
- 89 J. C. Martz, F. J. Freibert and D. L. Clark, *Nucl. Technol.*, 2021, **207**, S266–S285.
- 90 N. Harrison, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 4480–4485.
- 91 A. Svane, L. Petit, Z. Szotek and W. M. Temmerman, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 115116.
- 92 S. Heathman, R. Haire, T. Le Bihan, A. Lindbaum, M. Idiri, P. Normile, S. Li, R. Ahuja, B. Johansson and G. Lander, *Science*, 2005, **309**, 110–113.
- 93 A. Lukoyanov, A. Shorikov, V. Bystrushkin, A. Dyachenko, L. Kabirova, Y. Y. Tsiovkin, A. Povzner, V. Dremov, M. Korotin and V. Anisimov, *J. Phys.: Condens. Matter*, 2010, **22**, 495501.
- 94 K. Moore, G. van der Laan, R. Haire, M. Wall, A. Schwartz and P. Söderlind, *Phys. Rev. Lett.*, 2007, **98**, 236402.
- 95 L. Huang, R. Chen and H. Lu, *Phys. Rev. B*, 2020, **101**, 195123.
- 96 S. Heathman, T. Le Bihan, S. Yagoubi, B. Johansson and R. Ahuja, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 214111.
- 97 J. M. Sperling, E. J. Warzecha, C. Celis-Barros, D.-C. Sergentu, X. Wang, B. E. Klamm, C. J. Windorff, A. N. Gaiser, F. D. White and D. A. Beery, *Nature*, 2020, **583**, 396–399.
- 98 J. M. Sperling, E. Warzecha, B. E. Klamm, A. N. Gaiser, C. J. Windorff, M. A. Whitefoot and T. E. Albrecht-Schönzart, *Inorg. Chem.*, 2020, **60**, 476–483.
- 99 J. M. Sperling, N. Beck, B. Scheibe, Z. Bai, J. Brannon, D. Gomez-Martinez, D. Grödler, J. A. Johnson, X. Lin, B. M. Rotermund and T. E. Albrecht-Schönzart, *Chem. Commun.*, 2022, **58**, 2200–2203.
- 100 J. M. Sperling, E. Warzecha, C. J. Windorff, B. E. Klamm, A. N. Gaiser, M. A. Whitefoot, F. D. White, T. N. Poe and T. E. Albrecht-Schönzart, *Inorg. Chem.*, 2020, **59**, 10794–10801.
- 101 B. Xiao, M. Klinkenberg, D. Bosbach, E. V. Suleimanov and E. V. Alekseev, *Inorg. Chem.*, 2015, **54**, 5981–5990.
- 102 E. M. Villa, S. Wang, E. V. Alekseev, W. Depmeier, T. E. Albrecht-Schmitt, *Facile routes to ThIV, UIV, and NpIV phosphites and phosphates*, 2011, 3749–3754.
- 103 M. Qie, J. Lin, F. Kong, M. A. Silver, Z. Yue, X. Wang, L. Zhang, H. Bao, T. E. Albrecht-Schmitt and J.-Q. Wang, *Inorg. Chem.*, 2018, **57**, 1676–1683.
- 104 G. L. Murphy, Y. Wang, P. Kegler, Y. Wang, S. Wang and E. V. Alekseev, *Chem. Commun.*, 2021, **57**, 496–499.
- 105 G. L. Murphy, P. Kegler, M. Klinkenberg, A. Wilden, M. Henkes, D. Schneider and E. V. Alekseev, Incorporation of iodine into uranium oxyhydroxide phases, *Dalton Trans.*, 2021, **50**, 17257–17264.
- 106 G. L. Murphy, P. Kegler, E. M. Langer and E. V. Alekseev, *Crystals*, 2021, **11**, 965.
- 107 E. M. Langer, O. Walter, J.-Y. Colle, D. Bosbach and E. V. Alekseev, *Inorg. Chem.*, 2018, **57**, 1604–1613.
- 108 P. Toulemonde, C. Darie, C. Goujon, M. Legendre, T. Mendonca, M. Alvarez-Murga, V. Simonet, P. Bordet, P. Bouvier and J. Kreisel, *High Pressure Res.*, 2009, **29**, 600–604.
- 109 Y. Zhang, C. Zang, H. Ma, Z. Liang, L. Zhou, S. Li and X. Jia, *Diamond Relat. Mater.*, 2008, **17**, 209–211.
- 110 D. Costin, A. Mesbah, N. Clavier, N. Dacheux, C. Poinssot, S. Szenknect and J. Ravaux, *Inorg. Chem.*, 2011, **50**, 11117–11126.
- 111 H. R. Hoekstra and L. H. Fuchs, *Science*, 1956, **123**, 105–105.
- 112 A. T. Chapman, G. W. Clark, D. E. Hendrix, C. S. Yust and O. B. Cavin, *J. Am. Ceram. Soc.*, 1970, **53**, 46–51.
- 113 S. G. Duyker, V. K. Peterson, G. J. Kearley, A. J. Studer and C. J. Kepert, *Nat. Chem.*, 2016, **8**, 270–275.
- 114 N. Campagnol, T. Van Assche, T. Boudewijns, J. Denayer, K. Binnemans, D. De Vos and J. Fransaer, *J. Mater. Chem. A*, 2013, **1**, 5827–5830.
- 115 L. Paseta, G. Potier, S. Sorribas and J. Coronas, *ACS Sustainable Chem. Eng.*, 2016, **4**, 3780–3785.
- 116 D. Gui, W. Duan, J. Shu, F. Zhai, N. Wang, X. Wang, J. Xie, H. Li, L. Chen, J. Diwu, Z. Chai and S. Wang, *CCS Chem.*, 2019, **1**, 197–206.
- 117 Y. Wang, Y. Li, Z. Bai, C. Xiao, Z. Liu, W. Liu, L. Chen, W. He, J. Diwu, Z. Chai, T. E. Albrecht-Schmitt and S. Wang, *Dalton Trans.*, 2015, **44**, 18810–18814.
- 118 Y. Wang, X. Yin, J. Chen, Y. Wang, Z. Chai and S. Wang, *Chem. – Eur. J.*, 2020, **26**, 1900–1905.

