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Models of charge pair generation in organic solar cells

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Efficient charge pair generation is observed in many organic photovoltaic (OPV) heterojunctions, despite nominal electron–hole binding energies which greatly exceed the average thermal energy. Empirically, the efficiency of this process appears to be related to the choice of donor and acceptor materials, the resulting sequence of excited state energy levels and the structure of the interface. In order to establish a suitable physical model for the process, a range of different theoretical studies have addressed the nature and energies of the interfacial states, the energetic profile close to the heterojunction and the dynamics of excited state transitions. In this paper, we review recent developments underpinning the theory of charge pair generation and phenomena, focussing on electronic structure calculations, electrostatic models and approaches to excited state dynamics. We discuss the remaining challenges in achieving a predictive approach to charge generation efficiency.

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1 Introduction

Organic heterojunction devices have attracted intense interest for low cost photovoltaic applications, with record power conversion efficiencies for a single junction device rising from ~3% in 2001 to 11.1% in 2011.¹ Their function differs from that of inorganic p–n or p–i–n structures in that photocurrent generation depends on two stages: the separation of a photo-generated exciton into independent charge carriers, in addition to the successful transport of independent charge carriers to the electrodes. We do not attempt to give a comprehensive review of experimental literature here, as that would be beyond the scope of a single paper. Comprehensive and up-to-date reviews of experimental studies can be found in ref. 2–4.

The process of charge generation is widely believed to consist of the dissociation of the exciton to form an intermediate state involving bound charges at the interface between donor and acceptor materials, sometimes referred to as a charge transfer (CT) state, followed by separation into independent charges. One surprising observation is that, even though a simple coulombic treatment, assuming electrons and holes are point charges, a relative dielectric constant (ϵ_r) of 3–4, and a separation (r_{eh}) of 1–2 nm at the interface, results in a coulombic binding energy of ~0.1–0.5 eV (eqn (1)), significantly above the average thermal energy at room temperature, $E_{\text{Thermal}} \sim 0.025$ eV, many experimental

studies have found charge separation efficiencies approaching 100%.^{5,6}

$$E_{\text{Binding}} = \frac{e^2}{4\pi\epsilon_0\epsilon_r r_{eh}} \quad (1)$$

In modelling the process of charge separation, it has been common to distinguish the processes of exciton dissociation, and charge pair separation. The first part concerns the nature of the transition from photoexcited exciton (S^*) to the CT state (Fig. 1), and how this is controlled by energy levels of different blend components, while the second part concerns the nature of the transition from CT state to a charge separated state (CS, Fig. 1), and the material parameters that are important for determining this. In this section we introduce, and collect experimental evidence for, a number of proposed mechanisms for these processes.

It is widely held that the offset in energy levels between donor and acceptor material provides the driving force for the $S^* \rightarrow$ CT process,^{7–10} but there is some uncertainty around the energies and spatial extents of states involved, the sequence of transfer processes, and their dynamics.

A number of works have suggested that there is a threshold in this energy offset, ΔE_{CS} , between the singlet and CT state for efficient charge generation to occur. This concept is supported by work by Ohkita *et al.* showing a correlation between charge generation efficiency and free energy in a series of thiophene based polymers blended with PCBM,¹¹ Veldman *et al.*⁹ on a series of polymer:polymer and polymer:small molecule blends, and Faist *et al.*¹⁰ on a series of conjugated polymers blended with fullerenes of different acceptor strength. A number of low

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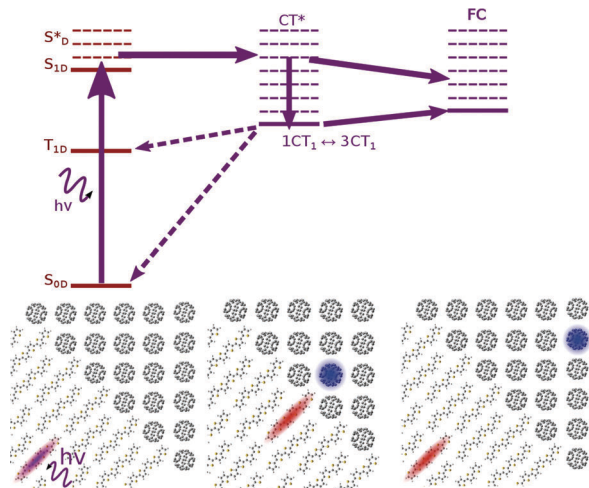


Fig. 1 Relevant states in the generation of charges in an organic photovoltaic device. An exciton is generated (typically in the donor) by photonic excitation ($S_0 \rightarrow S^*$). This exciton then migrates to the interface between donor and acceptor. Here, the electron transfers from the LUMO of the donor to the LUMO of the acceptor material to form a CT state (CT^*). This CT state must dissociate into a charge separated (CS) state, before or following thermal relaxation to CT_1 (at an energy E_{CT}), in order for an external current to flow. Dashed purple arrows indicate possible recombination pathways through which an excitation may be lost prior to the formation of a CS state.

bandgap isoindigo^{12,13} and diketopyrrolopyrrole¹⁴ containing polymers have subsequently been reported, which exhibit high charge separation efficiencies when blended with fullerene despite small energy level offset. It may be important that the low bandgap component is the polymer, rather than the fullerene, in these latter cases.

Driving forces for charge separation may also result from factors associated with the change in medium.

The delocalisation of charge carriers over a molecule (Fig. 2a) or a number of molecules (Fig. 2b) may be important in driving charge separation. Measurements on polycrystalline octothiophenes films by Knupfer *et al.* suggests a hole delocalisation over four to five thiophene units (~ 2 nm),¹⁵ and earlier work on P3HT in solution by Holdcroft estimates delocalisation over at least ten units (~ 4 nm).¹⁶ Shimoi *et al.* conclude from theoretical and experimental studies that hole polarons delocalise over 4 repeat units (~ 2 – 3 nm) in MDMO-PPV.¹⁷ If these polaron sizes are representative of charge distributions close to the interface, then this would result in a large effective separation of electron and hole, and have a significant impact on the coulombic barrier to charge separation. Recent studies by Bernardo *et al.*,¹⁸ and Savoie *et al.*¹⁹ also suggest delocalisation of the electron over multiple fullerenes may be important in determining the dependence of charge transfer state energies on fullerene content.

Another consideration is the possible involvement of electronically or vibronically excited charge transfer states (CT^* in Fig. 1), whereby any additional electronic and/or vibrational energy that an exciton possesses above the energy of the lowest CT state (E_{CT} , energy of CT_1 in Fig. 1) may be important in bringing about charge separation. A potential mechanism is the

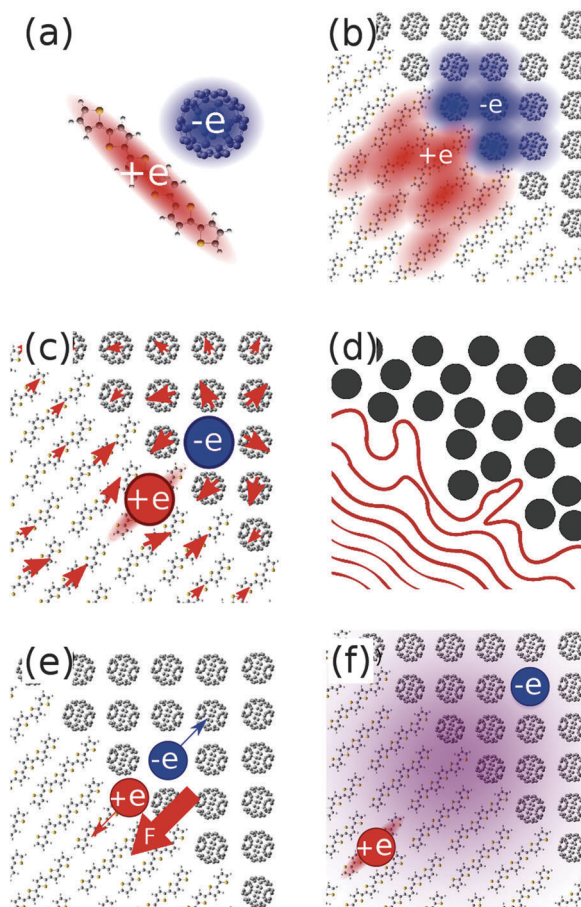


Fig. 2 Schematics of proposed mechanisms for efficient charge generation in OPVs. (a) Intramolecular or (b) intermolecular delocalisation of charge may result in a higher effective electron–hole separation, and reduced Coulomb binding in the charge transfer state, (c) a nonlinear molecular/atomic scale treatment of dielectric effects of the surrounding medium may result in changes to the energetic landscape for charge separation, (d) disorder at the interface may drive charges into single component domains, (e) the built-in electric field may drive electrons and holes apart at the interface, and (f) the initial excitation may be well delocalised, and directly form well separated charges without proceeding via an interfacial CT state.

coupling of higher lying excitons to more delocalised, higher energy charge transfer states. The existence of such delocalised excited CT states is supported by infrared (IR) pump-push measurements, showing generation of additional charges when a device containing relaxed CT states is excited with an IR pulse.²⁰ This concept is also supported by electroabsorption studies showing a rising electroabsorption signal over tens of femtoseconds to hundreds of picoseconds following photonic excitation. This rise is attributed to a growing electric field due to the separation of interfacial charge pairs, *via* charge transfer states with increasing degrees of charge delocalisation.^{21,22} However, recent studies show that the internal quantum efficiency of photocurrent generation is insensitive to photon energy over the range of photon energies spanning CT and single molecule excitations.⁶ Thus the importance of hot excitations for photocurrent generation is in dispute.



Microscopic treatments of electrostatics of the medium, which account for the effects of chemical structure, position and orientation of molecules at the interface (Fig. 2c), show that these can have a large impact on the magnitude and separation dependence of the charge pair binding energy in ways not captured by a linear dielectric constant (eqn (1)). This is largely due to anisotropic polarisabilities and charge distributions of considered molecules.

Disorder at the interface may give rise to a gradient in the energies of available electron or hole states, promoting charge separation (Fig. 2d). This concept is supported by a number of theoretical studies.^{23–25} Related to this is the role of entropy in driving separation of charges. There exist many states in which polaron pairs are well separated, and relatively few interfacial charge transfer states. As such, the separation of charges is entropically favourable.⁴

The built in electric field in a device at short circuit may influence charge separation efficiency, by driving both electron and hole toward opposite electrodes, and, in some cases, away from the interface (Fig. 2e). Experimental studies of devices using measurements of charge density and lifetime²⁶ or using a time delayed collection method^{27,28} show that charge pair generation can be enhanced by an applied electric field in the case of some relatively amorphous polymer:fullerene blends, though not in other blends. Another study by Veldman *et al.* found a slightly enhanced dissociation probability in PF10TBT:PCBM devices at typical operating voltages over short circuit.⁸

Another possible mechanism is the direct generation of free charge carriers from excitons located deep in the polymer domain, without proceeding *via* an intermediate interfacial CT state (Fig. 2f, bypassing the intermediate CT state in the central panel of Fig. 1), relying on a relatively slow decay of the excitonic wavefunction through the polymer domain.²⁹ A different mechanism which may avoid trapping at the CT state is direct photonic excitation to a superposition of energy eigenstates of different degrees of charge transfer, which, through lattice interactions, may couple directly to charge separated states.³⁰ This concept is supported by the ultrafast (~ 100 fs) generation of charges reported in transient absorption studies on a range of polymer:fullerene blends (P3OT:C₆₀,³¹ P3HT:PCBM,^{32–35} MDMO-PPV:PCBM,^{34–36} PFODTBT:PCBM,³⁷ PCPDTBT:PCBM,^{38–40} PCDTBT:PCBM^{34,35,41–43}), and in time-resolved resonance-Raman showing hole polarons generated far from corresponding electron polarons on timescales of 300 fs in PCDTBT:PCBM.⁴⁴

Experimental studies have yet to determine which mechanism is, or combination of mechanisms are, dominant in the generation of charges following photoexcitation. In the following section, we describe a number of phenomenological models which have been developed in an effort to understand and model the processes driving charge generation in OPVs. These studies contain little or no chemical information which can help us to differentiate between different materials systems. They do, however, motivate the study of these phenomena in specific material systems. The remainder of this review describes attempts to calculate the parameters relevant to different charge separation models, organised

around three key themes: electronic states at the interface and delocalisation of charge carriers, electrostatic effects, and dynamics of charge pair generation.

2 Early models

During the early stages of the development of theory for OPV device function, a number of phenomenological models were developed in an effort to represent the processes driving charge generation in OPVs.

One benchmark is Onsager's analytic model for ionic separation in weak electrolytes,⁴⁵ as adapted by Braun for the study of organic solids containing donor and acceptor units.⁴⁶ Mihailetschi *et al.* adapt this model to consider charge generation in organic photovoltaics. They are able to reproduce current density–voltage curves for polymer:fullerene solar cells of different composition, but only using an unphysically slow charge recombination rate of $1 \mu\text{s}^{-1}$.⁴⁷

To explain realistic charge yields, this model has been extended in a variety of ways. Peumans and Forrest's model considers the possible implications of the conversion of excess electrical energy of an exciton incident on the interface to kinetic energy, resulting in an initial separation of electron and hole at the interface.⁴⁸ This approach reproduces current–voltage curves, but only if assuming a large (48 Å) initial electron–hole separation, and a high attempt-to-jump frequency relative to charge recombination rates.

Deibel *et al.*⁴⁹ use a kinetic Monte Carlo device model to consider the impact of reduction in Coulomb barrier associated with a hole whose charge is delocalised over a number of monomer units, similar to the situation in Fig. 2a. They find a large increase in dissociation yield on increasing conjugation length between 1 and 10 monomer units, indicating that charge carrier delocalisation could have a very significant effect on dissociation probability.

In a contrasting approach, Arkhipov *et al.* consider the increase in the kinetic energy of the hole when coulombically localised by an electron on a fullerene, and positive charges on surrounding donor units (resulting from ground state partial electron transfer between molecules along the heterojunction).⁵⁰ This model relies on a large degree of groundstate electron transfer from polymer to fullerene ($\sim 0.1e$ per fullerene) and a very small effective mass of $\leq 0.3 m_e$ along the polymer chain, thought to be inconsistent with small electronic bandwidth in organic materials. More recently, Nenashev *et al.* combined the models of Deibel and Arkhipov, concluding that both effects may play an important role in determining charge dissociation probability.⁵¹

Offermans *et al.* study the impact of disorder in site energies on charge separation in a kinetic Monte Carlo simulation, finding that a difference in levels of disorder between the two materials could be important in driving separation of charges, *via* relaxation of hot charges near the interface into deeper energy sites that generally lie further away.⁵² A recent study by van Eersel *et al.* similarly concludes that disorder may play an important role in dissociation of charges at the interface.⁵³



Whilst many of the physical factors that may influence the rate of charge separation are invoked in these models, the models cannot relate processes in different materials to their chemical structure. For this, a means to calculate the electronic structure of the materials is required.

3 Electronic states at the interface

Most theoretical efforts to understand the process of charge separation have focussed on the energy and nature of the electronic states at the donor:acceptor interface. Electronic structure calculations of donor:acceptor combinations have the potential to relate observations to the specific chemical and physical structure of the molecules concerned. In most cases quantum chemical methods are used to study a single oligomer or molecule of the donor material and a single molecule of the acceptor. Properties of the electronic states of these molecule pairs are taken to be representative of states at the interface in a continuous system. These may play a role in the charge generation process, either as intermediaries between the exciton and free charges, or as trap sites at which excitons may recombine.

We will discuss the development of appropriate electronic structure methods separately to results on specific systems.

3.1 Electronic structure calculation methods

Calculations on excited interfacial states exhibiting charge transfer present theoretical and computational challenges. A balance must be struck between using sufficiently high levels of theory to correctly describe physical processes, and choosing system sizes which are large and detailed enough to be representative of relevant parts of an organic photovoltaic device. In this section, we give a brief summary of the challenges and methods involved. Factors which tend to increase and decrease calculated excitation energies to states exhibiting significant electron-hole separation are summarised in Fig. 3.

Density functional theory (DFT), whereby the interacting array of electrons are represented by a single electron density function, has enjoyed a great deal of success in reproducing properties of systems in which electrons are relatively delocalised. The Hartree-Fock method represents an alternative approach to electronic structure calculations, whereby exchange interaction between electron states is explicitly considered, but coulombic interactions are only considered *via* the mean field approximation. For organic systems, in which electrons are relatively tightly bound to molecules, hybrid functionals such as B3LYP,⁵⁴ containing an

empirical mixing of DFT and Hartree-Fock, have been very widely used.

Whilst methods for calculating ground state properties of chemical systems are relatively mature, the development of methods for calculating their excited states remains very much an active field.

A popular method for calculation of excited states is linear response time dependent density functional theory (TDDFT), which has been widely used to model excited states of organic molecules,⁵⁵ and is attractive for its computational efficiency and availability as part of many quantum chemical packages.

Unfortunately, when used with standard functionals, linear response TDDFT does not reproduce sufficient electron-hole binding energy.^{56–58} This results in an improper delocalisation of the excited state, and spuriously low charge transfer state excitation energies. These problems may be alleviated by the mixing of a larger component of Hartree-Fock (HF) exchange into the functional, but this will result in an improper description of local excitations, due to a lack of dynamic correlation.

In response to this problem, a number of long-range corrected functionals have been developed, in which the mix of HF and DFT depends upon the spatial separation.^{59–62} Yanai *et al.*'s Coulomb attenuating method (CAM-B3LYP),⁶² has proved popular, and has been shown by Peach *et al.* to well reproduce high level theory calculations for a series of small molecules exhibiting varying degrees of charge transfer.⁶³ More recently, tuned range separated hybrid functionals such as the Baer-Neuhauser-Livshits functional (BNL) have been developed,^{64,65} whereby the form of the attenuation factor is explicitly tuned to the system (or systems) under consideration.

Many-body methods beyond linear response and density functional theory should offer greater predictive power, but are currently only tractable for relatively small systems, or in conjunction with other broad approximations. An example of such an approach is the use of Green's functions equations of motion, containing both the nonlocal, energy-dependent electronic self-energy, and the electron-hole interaction leading to the formation of excitons, described by the Bethe-Salpeter equation (BSE).^{66–71}

The delta-SCF method, in which the HOMO of the donor is fit to the ionisation potential (IP(D)) and the LUMO of the acceptor to the electron affinity (EA(A)), represents another method of calculating accurate excited state orbitals, but is also relatively expensive.⁷²

Singles configuration interaction (SCI) represents a method of accurately calculating excited state energies, but is prohibitively expensive when used with standard functionals. SCI has, in a number of studies, been used with intermediate neglect of differential overlap (INDO) in order to study larger systems.⁷³ As charge is likely to delocalise between molecules, the neglected differential overlap between electron wavefunctions localised on different atoms implicit in INDO may be important in defining characteristics of CT states.

Constrained density functional theory (CDFT) represents another method of accessing charge transfer state properties at reasonable computational cost.⁷⁴ Here, donor and acceptor

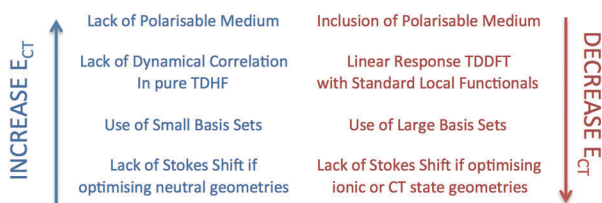


Fig. 3 Factors which will tend to increase and decrease calculated charge transfer excitation energies.



molecules are confined to have charges of +e and -e, and an excess spin of $\pm\frac{1}{2}$, and a self-consistent ground state DFT calculation is made. This allows the CT ground state to be engineered from a functional that would mispredict the charge localisation due to self-interaction error. CDFT is unable to predict states exhibiting partial electron transfer, and offers no direct route to higher excited states of the system.

Irrespective of functional, correct description of long range charge transfer requires a sufficient basis set to allow for orbital density at intermediate locations in space.⁷⁵

When calculations on a donor:acceptor molecular pair are to be used to represent the extended binary film, the electrostatic response of the surrounding medium needs to be accounted for. The surrounding molecules will, in general, become polarised by the charge pair and may contribute to the barrier to charge separation. Including the detailed response of the medium is a complex problem. A simple approach is to treat the medium as a continuum and carry out an electronic structure calculation in a spherical cavity within a continuous polarisable medium.⁷⁶

Going beyond the linear response of a dielectric, the surroundings can be modelled at the level of a polarisable empirical force field. Here a full self-consistent calculation of the electronic structure and microscopic polarisation of the medium is needed (potentially even including molecular reorientation).^{76,77} Such approaches, in which a quantum mechanical (QM) calculation is calculated, embedded within a classical molecular mechanical (MM) surroundings, are often referred to as QM/MM calculations.

There exists no gold standard of calculation which can describe the energies and nature of excited states in the donor:acceptor blend correctly in all cases. However, a number of studies have been carried out using subsets of these methods in order to address various aspects of charge generation in organic photovoltaics. Results of such studies are reviewed in the next two sections.

3.2 Bimolecular interfacial electronic states

First, we review studies that have used the electronic states of donor:acceptor molecular pairs as a model of the OPV charge separating interface.

An early work by Kanai and Grossman studied P3HT:C₆₀ with periodic boundary conditions, with four thiophene units and one fullerene molecule in each unit cell.⁷⁸ Kanai and Grossman use pure DFT for this study, and take unoccupied Kohn-Sham orbitals to represent excited states. The method is flawed because pure DFT is known to delocalise charge to an unrealistic extent in conjugated organic materials,⁷⁹ and because the Kohn-Sham orbitals are not formally connected to the excited states.⁸⁰ Nonetheless, the ideas and results presented have stimulated and informed further study in this area. The study found excited electron states localised on P3HT, fullerene, and 'bridge' states delocalised over both molecules, resulting from the hybridisation of a P3HT π^* state with a triply degenerate unoccupied state of the fullerene. Kanai and Grossman propose that such states may act as intermediate states to facilitate the observed ultrafast charge transfer in this system.

Huang *et al.*⁷³ used hybrid DFT (B3LYP/6-31g) ground states, with excited states by INDO/SCI to probe excitation energies, Coulomb interaction energies, radiative lifetimes, degree of charge transfer, and magnitude and orientation of the transition dipole moment of PFB:F8BT and TFB:F8BT monomer pairs.

This study found a strong dependence of excited state properties on alignment of the monomer pair. The authors report an attractive configuration in which electron accepting benzothiadiazole (BT) units of F8BT are aligned with electron donating triarylamine groups of PFB, resulting in an excited state of charge transfer character, and a repulsive configuration in which these units are no longer aligned, resulting in a state of excitonic character on the F8BT acceptor, in which the excited state is calculated to have a significantly shorter radiative lifetime. They also find intermediate 'exciplex' configurations, in which excitations exhibit a mixed charge transfer and excitonic character, with intermediate radiative lifetimes. This study has been reviewed in detail elsewhere.⁸¹

For some systems, in which energy levels of donor and acceptor LUMO are well separated, a spectrum of charge transfer states have been calculated at energies below that of the first single molecule excitation. These include higher charge transfer states in which the hole is more delocalised, resulting in a weaker Coulomb binding between electron and hole. Bakulin *et al.* have found such states in P3HT:PCBM and P3HT:F8TBT, where the polymers are modelled as oligomers using INDO/SCI.²⁰ Few *et al.*⁸² observed a similar trend in the excited states of dodecothiophene:PCBM pairs using TDDFT with B3LYP/6-31g*. The relative delocalisation of higher lying CT states supports the notion that excess exciton energy may help drive charge separation.

Due to the improper treatment of Coulomb interaction exhibited by TDDFT, this method cannot be used to study separation dependence of CT state properties. Whilst Coulomb's law will be reproduced by INDO/SCI, the neglected differential overlap of wavefunctions may be important in determining CT state properties.

We discuss two approaches that are capable of probing separation and orientation dependent properties. These works use a high level of theory to calculate properties of relatively small systems. Isaacs *et al.* use BNL, a tuned range separated hybrid functional, to calculate excited state properties of molecule pairs of C₆₀ with a series of boron(subphthalocyanine) molecules.⁷² Range separation parameters are tuned to fit the HOMO of the donor to the ionisation potential, and the LUMO of the acceptor to the electron affinity, as calculated by delta-SCF. Baumeier *et al.* use many-body Green's functions with the BSE to examine the excited state spectrum of a dicyanovinyl-substituted quaterthiophene:C₆₀ molecule pair.⁶⁹

In the work of Isaacs *et al.*, energetics of CT states are dominated by electrostatics, with low energy CT states where fullerene localises close to the positively charged thiophene unit in the CT state, and higher energies when the fullerene localises further from this unit. They find a difference in Coulomb binding for different molecular configurations of between 0.2 and 0.6 eV for fullerene with differently functionalised



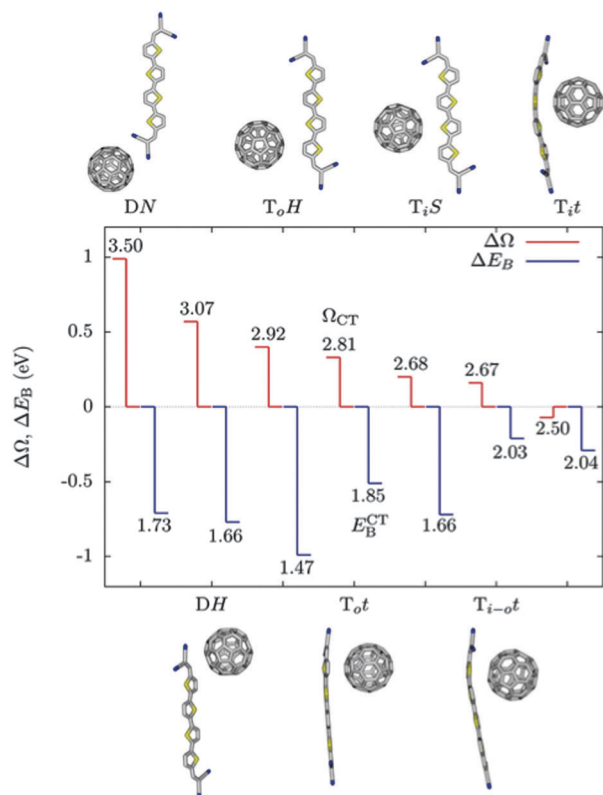


Fig. 4 GW-BSE calculations carried out by Baumeier *et al.*⁶⁹ on C₆₀:terminally substituted quarterthiophene molecule pairs. Difference between excitation energy to the CT state and S₀ → S₁ of the donor, $\Delta\Omega = \Omega_{CT} - \Omega_D$, in red, and reduction in binding Coulomb binding energy of the CT state relative to the ionisation potential of the donor, $\Delta E_B = E_B^{CT} - E_B^D$, in blue. The numbers are the excitation energies, Ω_{CT} , to the CT state, and the Coulomb binding energies, E_B^{CT} , of the CT state. All energies in eV. Molecular arrangements with higher excitation energies tend to have a lower E_B^{CT} , and more negative values of ΔE_B , indicating a larger reduction in the Coulomb barrier to charge separation relative to the ionisation potential of the donor.

boron(subphthalocyanine) molecules. Baumeier *et al.* find a similar position dependence of CT state energetics (Fig. 4), with E_{CT} lower by ~ 1 eV when the fullerene is localised close to the thiophene compared to when the fullerene is localised close to the terminal dicyanovinyl group. Changes in Coulomb binding energy of electron and hole in the CT state, calculated by summing intermolecular Coulomb interactions between atomic partial charges, do not exactly match changes in E_{CT} . This is attributed largely to the role of molecular polarisation in distorting orbitals in response to the presence of a charged molecule. In the DN configuration, a state is calculated in which 0.75 electrons are transferred from donor to acceptor, attributed to the proximity of C₆₀ to the strongly electronegative dicyanovinyl group, and to the close proximity of two units with acceptor character, making it an outlier in Fig. 4.

These studies suggest that the role of electrostatic interaction is more important than the overlap of orbitals in defining the energetics of interfacial excited states. This suggests that the influence of the environment could be very important in determining charge transfer state energetics. However, in cases where LUMO energies of the two materials are quasi-degenerate,

differences in coupling between different molecular arrangements can have a large impact on interfacial states, and potentially on charge generation efficiency. In such cases, small changes in chemical structure can also have a dramatic effect on the energy and nature of calculated excitations.

Use of a lower level of theory allows the calculation of electronic states of larger molecule pairs at reasonable computational cost. Few *et al.* apply linear response TDDFT with the global hybrid functional B3LYP/6-31g* *in vacuo* to the calculation of excited states of a wide range of similarly oriented conjugated oligomer:fullerene molecule pair. They show this method to well reproduce trends in electroluminescence energies from corresponding polymer:fullerene blends (Fig. 5a). The success can be partly attributed to the cancellation of errors due to the use of linear response TDDFT (tending to reduce the calculated excitation energy to charge transfer excitations, Fig. 3), and the lack of polarisable medium, relatively small basis sets, and lack of Stokes shift (tending to increase the calculated excitation energy to charge transfer excitations, Fig. 3), and these errors being sufficiently consistent between similar materials to establish meaningful trends.

Few *et al.*⁸² apply the same calculation method to a range of polymer:fullerene systems, and investigate the effect of ring alignment (along a few central oligomer units) of the fullerene on the nature and energetic of the excited state spectra of the molecule pairs. CT state properties exhibit only a weak

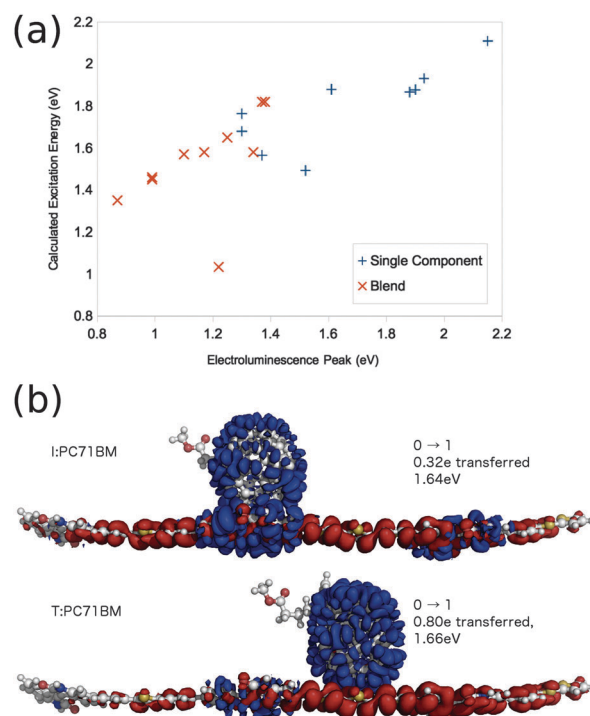


Fig. 5 Few *et al.*'s⁸² TD-B3LYP/6-31g* results on oligomer:fullerene pairs. (a) Calculated molecule pair, and single molecule excitation energies, plotted against electroluminescence energies measured for corresponding polymer or polymer:fullerene blend. (b) Calculated first excitation of a 33TI:PC71BM molecule pair. Electron density moves from red to blue regions in going from the ground to the excited state.



dependence on ring alignment of the fullerene and oligomer for pairs of molecules with well separated LUMO levels, but a much stronger dependence for molecule pairs with near degenerate LUMO levels. In particular, in P3TI:PC71BM, a system in which efficient charge generation is reported despite very close lying LUMO levels of donor and acceptor,^{12,13} the degree of electron transfer from donor to acceptor molecule varies from 0.32e to 0.80e for a fullerene aligned with different subunits on the central monomer (Fig. 5b). This is attributed to hybridisation of orbitals when fullerene is located close to units on the oligomer where the LUMO is heavily localised. For a number of chemical analogues of PDPP-TT-T alongside PC₇₁BM, another molecule pair with a small LUMO–LUMO offset,¹⁴ small changes in chemical structure have a large impact on charge generation efficiency. Few *et al.* calculate the degree of electron transfer exhibited in the first excited state for oligomers of these analogs with PC₇₁BM, and a correlation is found between systems calculated to exhibit a large degree of electron transfer in the first excited state, and those which efficiently generate charges in a device.

In summary, the energies and nature of excited states at the donor:acceptor interface can be explored with reasonable accuracy using available quantum chemical methods. The cheapest methods are based on DFT, and whilst these methods suffer from known limitations, the methods have proved useful in probing effects of the chemical structure, separation and orientation of molecular pairs on interfacial states. Probing the electronic structure of states that are delocalised over the molecular assembly is more expensive, and whilst experimental evidence suggests the delocalisation of states is important, only limited computational studies have been made so far.

3.3 Electronic states in larger arrays of molecules

The studies discussed in the previous section concern excited states of an individual donor:acceptor molecule pair. Such studies cannot evaluate the role of intermolecular delocalisation on charge separation. The concept of delocalised states is relevant to a number of studies that have shown that the well established redshift of CT state emission with increasing fullerene content in organic solar cells,^{18,83–85} can be explained partly in terms of fullerene crystallisation. Higher ratios of fullerene have also been shown to correlate with higher charge generation yield, and with a larger photocurrent and PCE in devices.^{18,84,86,87}

Though clearly important to study the effect of state delocalisation for these reasons, calculating the degree of charge delocalisation over assemblies of molecules requires a large system size (to allow the charge to be non-local), and is a challenge for DFT due to the self-interaction error. Use of more coarse-grained semi-empirical methods can allow the study of such systems.

One such approach is taken by Cheung and Troisi, in order to study the electrical properties of an assembly of PCBM molecules in the absence of a molecular donor.⁸⁸ They carry out molecular dynamics on a $3 \times 3 \times 3$ supercell, initially in the PCBM structure crystallised from oDCB solution, as reported by Rispen *et al.*⁸⁹ Electronic structure calculations are then

carried out on two thousand simulation snapshots. Fullerene molecules are coarse grained. ZINDO is used to calculate the first three quasi-degenerate LUMOs of a single PCBM molecule in vacuum. A rigid structure as used in the ZINDO calculation is superimposed onto each molecule from the MD snapshot, and reference molecular orbitals are imposed onto this molecule. The overlap of the degenerate LUMOs of the PCBM molecules is calculated instead of directly evaluating the electronic coupling, and overlap is only considered between atomic orbitals within 7.2 Å. Electronic states of the system are then calculated using a tight binding model, with molecular orbital site energies, and transfer integrals as calculated by orbital overlap.

Cheung and Troisi define a localisation length as twice the standard deviation of the orbital density with distance from the expectation position for the orbital. The lowest excitations of the system are found to be localized on one or two fullerene molecules (localisation length ~ 12 Å), and stabilised by 0.08 eV relative to the single molecule LUMO. Higher states are delocalised over a large number of molecules, and some over the entire simulation space (localisation lengths ~ 30 – 50 Å), many of which are thermally accessible from the lowest excitations.

Savoie *et al.* apply a similar method to calculate electronic states of crystallites of PCBM of sizes ranging from $1 \times 1 \times 1$ to $4 \times 4 \times 4$ unit cells (4 to 256 molecules),¹⁹ also prepared in Rispen *et al.*'s crystal structure from oDCB solution.⁸⁹ Savoie *et al.* calculate electronic states for the crystallite in isolation (Fig. 6a and b), and when a positive point charge is placed alongside the crystallite (Fig. 6c and d). Whilst the lowest excited state is localised close to the positive charge (Fig. 6c), many thermally accessible states are calculated which are delocalised over a large number of fullerene molecules (Fig. 6d). In the case of the $4 \times 4 \times 4$ supercell, the effect of delocalising the

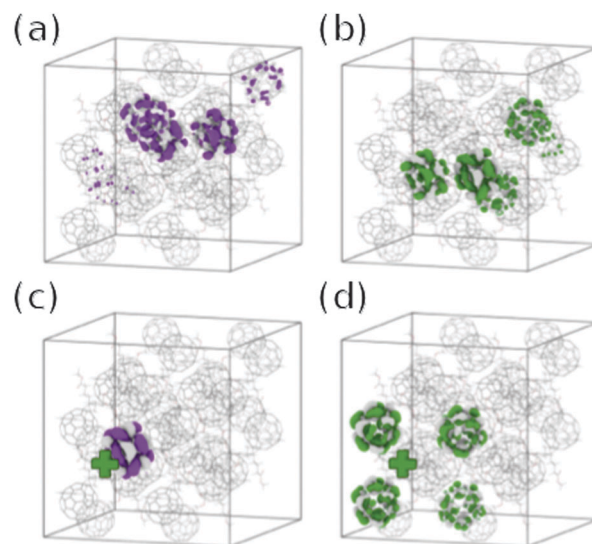


Fig. 6 States calculated by Savoie *et al.*¹⁹ in a $2 \times 2 \times 2$ crystallite of PCBM. (a) more localised, and (b) thermally accessible more delocalised electron state in the absence of a positive charge; and (c) more localised, and (d) thermally accessible more delocalised electron state in the presence of a positive point charge.



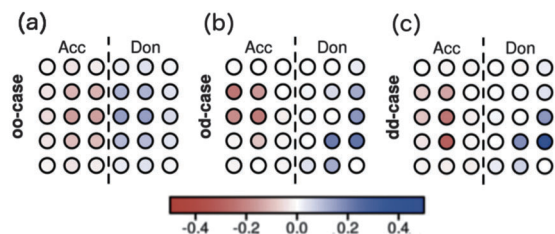


Fig. 7 Charges in the first excited singlet states from the CIS calculations on the two-dimensional hetero-junctions as calculated by Raos *et al.*⁹¹ for (a) no disorder, (b) disorder in couplings, but not in site energies, and (c) disorder in both couplings and in site energies.

positive charge uniformly in one or two dimensions is also considered, yielding a qualitatively similar set of electronic states.

This is interpreted as offering a possible pathway for efficient charge separation *via* higher lying, delocalised charge transfer states. Savoie *et al.* find that the average state localisation is mostly unaffected by rotational and translational disorder of PCBM molecules, attributing this to the spherical symmetry and resulting near-isotropic electronic coupling of the PCBM molecule. They ascribe the dominance of fullerene based acceptors in organic photovoltaics to this robustness of the fullerene electronic structure to structural disorder, a factor previously suggested to explain high mobilities in disordered C₆₀ films.⁹⁰ A comparative study on non-fullerene acceptors would therefore be interesting.

An interesting approach is taken by Raos *et al.*,⁹¹ who coarse grain the donor:acceptor interface as a regular grid of donor and acceptor molecules on a regular lattice. Each molecule is assigned only HOMO and LUMO orbitals, and transfer integrals with neighbouring molecules taken from a Gaussian distribution, weighted by a factor that decays exponentially with molecular separation distance. All values required as inputs to the Hamiltonian can be obtained from the ionisation energy, electron affinity, and singlet and triplet excitation energies of the molecule under consideration, which could be obtained from accurate gas phase measurements, or *ab initio* quantum chemical calculations.⁹²

Raos *et al.* apply this method to two-dimensional model heterojunctions of 15 donor and 15 acceptor sites. In the first excited singlet state of the ordered system, electron and hole are delocalised over several molecules (Fig. 7a). In the first excited singlet states with disorder in coupling (Fig. 7b) or disorder in both site energy and coupling (Fig. 7c), charge is transferred chiefly from molecules which are not directly adjacent to the interface. Similar results are obtained for a larger assembly of 66 donor and 66 acceptor sites.

The results of Raos *et al.* are obtained from only one system, using one set of site energies and transition integrals. Studies of sensitivity of results to parameter values and system configuration would be required in order to draw general conclusions.

4 Electrostatic effects

As mentioned in the introduction, one proposed explanation for efficient charge generation in organic photovoltaics is a classical screening effect, brought about by induced polarisation in

molecules surrounding the CT state. This effectively reduces the Coulomb interaction between electron and hole, and so possibly facilitates charge separation. In the limit of a homogeneous continuous medium, this screening of the Coulomb interaction *via* polarisation is included in eqn (1) through the ϵ_r term. However, a constant ϵ_r is only appropriate for materials whose electronic response can reasonably be considered linear, isotropic, and homogeneous. Whilst this may be a fair approximation for large scale systems such as capacitors where bulk effects are important, it does not accurately describe the polarisation of the medium surrounding a charge transfer state, which takes place on a molecular scale (Fig. 2c). The positions and orientations of molecules, along with their charge distributions and polarisabilities, can have a large impact on the energetic landscape for charge separation.^{24,93}

Such effects are extremely challenging to probe experimentally. This is, in part, due to differences in dielectric constant when probed at different frequencies. At high frequencies, nuclei may be assumed to be static, and the freedom of electron orbitals to rearrange in response to applied field will be the main factor defining the dielectric response. At lower frequencies, this assumption breaks down, and the reorientation of nuclei in response to applied field will typically result in a higher value for dielectric constant. Values for bulk permittivity at low frequencies can be deduced using capacitance measurements, but can be very sensitive to device architecture. Values of ϵ_r at optical frequencies can be obtained using ellipsometry, sometimes in different crystal directions.^{94–97} Which of these values is relevant to the screening experienced when charges separate, or indeed whether a single number (or tensor) dielectric constant is sufficient to describe this process, is unclear.

Still more challenging to probe are dielectric effects at the interface between two material systems. Permanent interfacial dipoles, either resulting from molecules with an intrinsic ground state dipole, or *via* ground state partial electron transfer between donor and acceptor molecules, may also play an important role in defining the energetic landscape for charges at the interface. Ultraviolet photoemission spectroscopic studies have been carried out measuring shifts in ionisation potential in a sample with a controlled number of molecular monolayers, suggesting some ground state charge transfer in a number of organic:organic systems relevant to organic photovoltaics.^{98–102}

Determining dielectric effects in general systems is a challenging problem, and not a new one. The Clausius–Mossotti relation, published in 1879, gives an exact expression for the dielectric constant of a simple cubic lattice of point dipoles of a given isotropic polarisability. This has been successful in describing some simple systems, but breaks down when considering disordered systems, systems in other crystal structures, or molecules whose polarisability is not well reproduced by an isotropic polarisability tensor. A number of schemes have been developed based upon the generalised Clausius–Mossotti (GCM) relation, whereby the system is modelled as a finite set of interacting polarisable dipoles, in the presence of an externally applied electric field (representing applied field and/or permanent charges and multipoles in the system).¹⁰³

The polarisable dipoles in the GCM equation have been chosen to represent units of different size in different schemes. A number of



schemes exist whereby molecules are broken down into atomic points, each of which is assigned a polarisability, and atomic polarisabilities are calibrated in order to reproduce those of molecules.^{104–109} These approaches use various forms of smearing of dipoles and/or separation cutoffs for dipole–dipole interaction to avoid infinite polarisation by the cooperative (head to tail) interaction between induced dipoles.

Polarisabilities of single molecules can sometimes be inferred from optical properties using the Clausius–Mosotti relations,^{104,105} or may be calculated using quantum chemical methods, although a careful choice of functional must be made when calculating polarisabilities of long conjugated molecules.^{65,110–114}

Electrostatic effects particular to interfaces may influence the energetic landscape for charge carriers in a number of ways, some of which are presented in Fig. 8.²⁴ Changes in dielectric constant will tend to stabilise charges in the ϵ_{low} domain close to the interface, while driving charges deeper in the ϵ_{high} domain (Fig. 8a). Poor packing may effectively result in a lower dielectric constant at the interface, driving charge carriers away (Fig. 8b). A static interfacial electric field due to multipole moments of organic semiconductors will result in a bending of bands close to the interface (Fig. 8c). Structural disorder at the interface, depending on its nature, may result in a range of effects. The impact on energy levels of disorder, whilst significant, is thus not easy to quantify (Fig. 8d).

Verlaak *et al.* use a GCM-like approach to probe an interface of C_{60} with the [0,0,1], and [0,1,−1] faces of a pentacene crystal.⁹³ Molecules are given molecular polarisability tensors, α_{mol} , and permanent quadrupole moments Θ_{mol} split between a number of sites, s , on the molecule each with polarisability α_{mol}/s , and quadrupole moment Θ_{mol}/s . The polarisation of each molecule is calculated in response to an electron–hole pair (represented as point charges) on various molecules in the system, and a value is obtained for the net electrostatic energy due to interactions of charges, induced dipoles, and quadrupoles is calculated.

By comparison of the energies of systems with electrons and holes close to and far from the interface, Verlaak *et al.* determine

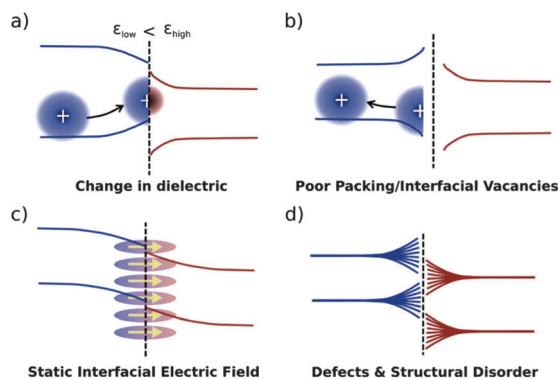


Fig. 8 Yost *et al.*'s²⁴ schematic representation of four different environmental effects on the organic/organic band structure: (a) a difference in dielectrics, (b) poor molecular packing at the interface, (c) a molecular multipole moment creating an electric field at the interface, and (d) a rough depiction of general disorder at the interface.

a barrier of only 0.04 eV for a pentacene [0,1,−1]: C_{60} interface, and 0.44 eV for a pentacene [0,0,1]: C_{60} interface. This is despite a closer approach of nearest neighbour molecules on the interface in the [0,1,−1]: C_{60} case. This is attributed to a greater stabilising interaction with induced dipoles for electron and hole further from the interface, coupled with a large driving force associated with changes in interactions between charges and permanent quadrupoles on the pentacene molecule. Linares *et al.* carry out further work on the pentacene: C_{60} junction using this approach,¹¹⁵ as reviewed elsewhere.⁸¹

These results are certainly interesting, however, there are some features of the model which warrant attention. The molecular polarisability tensor is split by simple division, resulting, in the case of pentacene, in atomic or ring polarisabilities which are far from isotropic. In reality, the anisotropic polarisability of pentacene must be due to the anisotropic arrangement of atoms and electrons, and heavily anisotropic polarisability tensors on each individual site may not give an accurate representation of the polar response on a submolecular scale. Secondly, electron and hole polarons are represented as point charges. Baumeier *et al.*⁶⁹ find very different energetic landscapes for interfacial charge pairs when considering charges as molecule centred points, and explicitly considering charge distributions.

Chia *et al.*¹¹⁶ model the dielectric response of crystallites of ~ 1000 fullerenes, in which each molecule is treated as a single polarisable point. A direction dependent dielectric constant is obtained by considering the energy surface associated with separation of charges along one or other of the crystal axes. Despite the simplicity of this approach, it has been able to well reproduce mean polarisabilities of C_{60} and PCBM.

QM/MM calculations also help to elucidate the role of electrostatics in determining behaviour of charges at the interface, by providing an estimate of the energetics of single QM molecules at different positions within an assembly of MM molecules, arranged to form a heterojunction.

Yost *et al.* apply such an approach to a number of organic semiconductor interfaces. They find an energy profile akin to Fig. 8a for a Rubrene (epsilon ~ 2.7): C_{60} (epsilon ~ 3.8). Pulling CuPc:PTCBI apart at an interface, they find an energy profile akin to Fig. 8b. Yost *et al.* examine an interface between C_{60} and a half unit cell of (dicyanomethylene)-2-methyl-6-(4-dimethylaminostyryl)-4H-pyran (DCM), which exhibits a significant dipole moment. They find a large bending of bands, similar to that shown in Fig. 8c, but to a greater magnitude such that the HOMO and LUMO orbitals of DCM are pushed below those of C_{60} close to the interface.

Poelking *et al.* find that long range interactions play a decisive role in the electrostatic landscape of a dicyanovinyl-substituted oligothiophene (DCVnT): C_{60} device.¹¹⁷ They perform an Ewald split, superimposing a non-periodic foreground onto a periodic, neutral background. They use Thole's model, based upon an interacting set of atomic polarisabilities, to model their foreground. They find that the inclusion of this background changes the energetic landscape from one which does not allow charge separation, to one which favours charge separation.



The geometrical relaxation of surrounding molecules may also play a role in determining the energetic landscape for charge carriers, and may be probed using molecular dynamics simulations. de Gier *et al.* study the stabilisation of a monomer: fullerene charge transfer state when surrounded by monomers with sidechains of various dipole moment, using molecular dynamics for initial configurations, and allowing sidechains to respond to the presence of the charge transfer state.¹¹⁸ They find a stabilisation of their charge pair by as much as 0.8 eV associated with sidechain orientation in the system with most polar sidechains, compared to 0.3 eV or less for less polar sidechains. This implies that the reorientation of sidechains could play a significant role in stabilising separated charges. However, it is not clear whether such processes could take place on the ultrafast (~ 100 fs) timescales associated with the onset of polaron absorption in spectroscopic studies. Such reorganisation may still impact the later generation of charges, or reduce nongeminate recombination.

In summary, it is widely accepted that the treatment of the dielectric environment as a homogenous material of known permittivity is a crude approximation that may lead to inaccurate estimates of the binding energy of charge pairs at the donor: acceptor interface. Microscopic electrostatic approaches, where individual molecules or parts of molecules are represented by polarisable units and the electrostatic energy calculated using a classical sum of coulombic, dipolar and quadrupolar interactions, have been used to investigate the situation at a variety of donor: acceptor interfaces. The resulting interaction energy depends strongly on molecular organisation at the interface as well as on chemical structure, and may differ in magnitude and sign from estimates based on the bulk ϵ_r . Attempts to include the effect of molecular dynamics on electrostatic binding energy have been made, though the relevance of these to charge separation depends on the relative rates of the electronic and molecular motions.

5 Dynamics of charge pair generation

All the works mentioned so far have focussed on modelling the energetic landscapes within an organic photovoltaic device. Whilst energetic landscapes are clearly important for defining characteristics favourable to charge generation, the process of generating charges is a dynamic one, and we are ultimately concerned with rates at which different processes take place. There exists no one accepted methodology for the calculation of rate processes in OPVs, but there have been a number of relevant publications calculating rates of processes on a bimolecular, and multimolecular scale.

In order for efficient charge generation, the rate of energy transfer from S_{1D} to CT_1 or CT^* must compete with spontaneous emission from S_{1D} back to S_{0D} , and the rate of separation of charges from CT_1 or CT^* to CS must compete with recombination rates from CT back to the ground state S_{0D} (Fig. 1).

Spontaneous emission rates may be computed directly from excitation energies and transition dipole moments of excited

states using Fermi's golden rule. These rates will give an indication of the recombination timescales with which charge separation must compete. Huang *et al.* take this approach in studying oligomer pairs.⁷³ They find spontaneous emission rates from states of chiefly single molecule excitation character on the order of 1 ns, and from states of exciplex or polaron pair character of ~ 0.1 – 10 μ s.

Under the frozen orbital approximation, rates for hopping between states on different molecules may be calculated from the coupling of orbitals, and the reorganisation energy associated with the movement of the charge by use of Marcus theory. Marcus theory is also based upon Fermi's Golden Rule, and, as such, is only valid under the approximation of weak coupling between states, which may not be valid for excited state dynamics in organic systems. It is also limited by difficulties in knowing the appropriate reorganisation energy for excited state transitions. By comparing transfer rates between calculated CT states, and ground and excited states on donor and acceptor, it may be possible to predict how electron transfer processes at the interface proceed.

An example of such an approach is Yi *et al.*'s¹¹⁹ calculations of couplings and hopping rates between a sexithiophene (6T) donor molecule, and C_{60} or perylenetetracarboxydiimide (PDI) acceptor molecule. Yi *et al.* represent the CT states of the donor: acceptor complex as a linear combination of ground states of the individual donor cation and acceptor anion, and calculate couplings with neutral ground and excited states (calculated using TDDFT) of each single molecule.

Yi *et al.* find that couplings are strongly dependent on mutual position of molecules. In most positions, couplings between CT state and ground state are stronger, and resultant recombination rates faster, for PDI ($\sim 10^9$ – 10^{12} s^{-1}) than C_{60} ($\sim 10^8$ – 10^{10} s^{-1}). This may be an important factor in explaining the more efficient charge generation in polythiophene: fullerene devices than polythiophene: PDI devices.

Difley *et al.* also use Marcus theory to calculate transition rates two donor: acceptor small molecule pairs, using TDDFT for single molecule states, but CDFT, in which donor and acceptor molecules are confined to have charges of +e and -e, and an excess spin of $\pm \frac{1}{2}$, to calculate charge transfer states, allowing electron and hole wavefunctions to be influenced by the presence of charges on the opposite molecule.

Use of molecular dynamics simulations allows comparison of relative rates for molecules pairs in a range of geometries. Liu *et al.* apply a Marcus rate approach to the calculation of charge separation and recombination rates for a P3MT (modelled as a hexamer):PCBM interface, finding charge separation rates on the order of 10^{10} – 10^{12} s^{-1} , and charge recombination rates between 10^6 – 10^9 s^{-1} , strongly dependent on mutual position of molecules. These results are comparable to experimentally reported charge separation rates faster than 4×10^{11} s^{-1} , and recombination rates between 10^8 and 10^9 s^{-1} .¹²⁰ These results suggest that the rate of charge transfer from the CT state to neighbouring units (CT to CS in Fig. 1) is faster than the relaxation from the CT state back to the ground state for a thiophene donor and PCBM acceptor. However, the relative rates of these processes are very



dependent upon mutual position and orientation of donor and acceptor molecules at the interface, and recombination rates from the CT state are sufficiently high that this could act as significant loss pathway, especially if charges are unable to escape from the vicinity of the interface.

It is also possible to go beyond rate based approaches, and perform quantum dynamics calculations. This approach requires solving the time dependent Hamiltonian for evolving states. An example of such an approach is Tamura *et al.*'s¹²¹ work, using the multiconfiguration time dependent Hartree (MCTDH) method to calculate dynamics of excited species in a quaterthiophene:C₆₀ molecule pair. They find that spatial separation of molecules has a large influence on dynamics at the interface, with the excitation oscillating between single molecule exciton and charge transfer state for a separation of 3.5 Å, and exciton/CT state population ratio settling down around one quarter after the rapid exciton decay at a separation of 3.0 Å, albeit still with large temporal fluctuations. Tamura *et al.* take this as evidence that the CT state has a degree of excitonic character, allowing a radiative decay to the ground state. Tamura and Burghardt have also applied this approach to larger assemblies of thiophene:C₆₀ pairs, finding charge separation to occur from an exciton close to the interface on a ~100 fs timescale (Fig. 9).¹²²

Some dynamic models are based on the possible generation of free charges without proceeding *via* intermediate interfacial states. This may occur either *via* long-range hopping following the localisation of the excitation on a single molecule, or due to an initially well delocalised state, or superposition of states, which directly decomposes into a state involving well separated charges. Both rely on the idea that photoexcited states, which may already be spatially delocalised, are coupled to charge separated states.

The case of long-range charge separation from a localised exciton has been modelled by Caruso and Troisi using a Marcus-based hopping model. This model is used to compare hopping rates from excited states deep within the donor

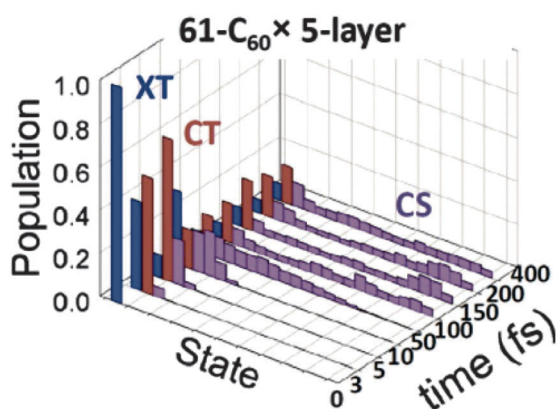


Fig. 9 Excitation dynamics from Tamura *et al.*'s¹²² MCTDH simulations of a stack of thirteen thiophene oligomers alongside five layers of 61 C₆₀ molecules. The initial excitation (XT), rapidly migrates to the interface to form a CT state, which decays to separated charges (CS) on a ~100 fs timescale.

domain to neighbouring donor molecules, and directly to charge-separated states. They assume an exponential decay of electron wavefunction over space, with a range of decay rates, justified by comparison with those reported for donor-bridge-acceptor molecules. Caruso and Troisi conclude that, for an exponential decay rate of the exciton wavefunction of less than 0.6 Å⁻¹, the majority of electron-hole pairs are generated without the exciton reaching the interface. This decay length is, however, longer than values found even for intramolecular charge transfer in many systems.^{123–125}

The case of charge separation without prior localisation of the exciton has been modelled by Bittner and Silva.³⁰ Here, the initial photoexcitation is to either an energy eigenstate, or a coherent superposition of energy eigenstates of the system, which evolve with time according to the time dependent Schrodinger equation. Decoherence is driven by interaction with a time dependent coupling between orbitals, brought about by environmental fluctuations. Bittner and Silva apply this model to a donor oligomer:acceptor oligomer heterojunction, each domain containing four oligomers with five possible electron sites, and add energetic disorder, in one case close to the interface, and in another, far from the interface. Calculated state energies *versus* charge separation distance are shown in Fig. 10. They find a fast transfer rate from initial excitation to free polaron states, suggesting that electron or hole could tunnel 12–15 Å from the site of the initial excitation within 35 fs. This is assigned to strong resonance between high oscillator strength excitons, and near-degenerate free polaron states, which are almost always present given a sufficient level of energetic disorder in the simulation space.

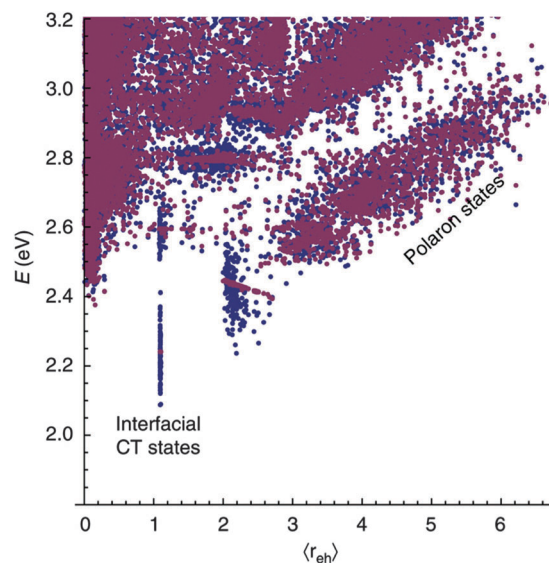


Fig. 10 Bittner and Silva's³⁰ calculated state energy *versus* charge separation distance, for energetic disorder close to the interface and order in regions away from the interface (blue), and energetic order close to the interface and disorder in regions away from the interface (purple). High oscillator strength excitonic states are calculated to exhibit resonance and fast transfer rates to near-degenerate (but low oscillator strength) polaron states.



Compared to the theoretical study of the energies of states at the donor:acceptor interface, the theoretical treatment of transitions between such states is much less well developed. Approaches based upon Fermi's Golden Rule have been used to quantify the rates of transitions between excitonic states, charge transfer states and ground, but are limited by the validity of the weak coupling approximation. More powerful approaches are based on the evolution of states within the time dependent Schrodinger equation and allow multiple state configurations to be studied. These approaches are promising, but still under development, and are likely to require large approximations, or be computationally expensive for representative systems.

6 Summary and concluding remarks

Whilst there is as yet no complete theory of charge generation at organic heterojunctions, different parts of the problem have been addressed and significant progress has been made in building theoretical tools and in rationalising experimental results. The work has clarified the directions for development of more powerful models.

High level calculations of molecule pairs *in vacuo* indicate that mutual position of donor and acceptor molecules is very important in defining the energies of CT states. The nature and energy of excited states are particularly sensitive to molecular alignment when the LUMO of donor and acceptor are close in energy. Semi-empirical calculations of states in larger arrays of molecules indicate that the states involved in charge transfer may delocalise over many molecules. This work indicates that more efficient treatments of excited states in multi-molecular systems are needed. However, this capability will also require improved experimental probes of the physical structure of interfacial regions.

Studies on electrostatics show that changes in dielectric constant, poor packing, static internal electric fields, and structural disorder can all influence the energetic landscape close to the heterojunction. The orientation of molecules, and alignment of crystals at the interface, can be decisive in determining this landscape, in a manner which could not be captured with a bulk dielectric constant. Long range interactions, and edges of the device can also play a defining role in the energetic landscape for charge separation. The significance of local structural detail again indicates the need for improved structural probes in order to apply the models usefully to practical situations.

Calculated transition rates on molecule pairs agree reasonably well with the range of those reported experimentally, and studies on the impact of mutual position and orientation of molecules may provide insights into favourable orientations for separation of charges. Explicit calculations on the evolution of states in larger arrays of molecules indicate the possibility of ultrafast separation, which may proceed *via* the CT state or directly to charge separated states. On longer timescales, the dynamics of molecules close to the interface may influence charge separation efficiency, but little progress has been made in this area so far.

Whilst these studies do not provide a unique mechanism for charge separation, they help to inform the roles that molecular properties may play in these mechanisms. They provide a greater understanding of the importance of orbital energies and polarisabilities of molecules, ways in which molecules may be arranged in order for couplings and energetics favourable for charge separation, the possible role of disorder, and relevant timescales for this process. Through further study, and with the aid of more detailed information on the structure and the ultrafast dynamics of charge transfer processes, the roles of these different mechanisms in driving the efficient charge separation observed in many OPV material systems may be better understood, and design rules developed for OPV structures of improved performance.

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References

- 1 National Renewable Energy Laboratory, *Best research cell efficiencies*, 2014.
- 2 M. Antonietta and A. Troisi, *Phys. Chem. Chem. Phys.*, 2014, **16**, 20277–20278.
- 3 F. Gao and O. Inganäs, *Phys. Chem. Chem. Phys.*, 2014, 20291–20304.
- 4 T. M. Clarke and J. R. Durrant, *Chem. Rev.*, 2010, **110**, 6736–6767.
- 5 S. H. Park, A. Roy, S. Beaupre, S. Cho, N. Coates, J. S. Moon, D. Moses, M. Leclerc, K. Lee and A. J. Heeger, *Nat. Photonics*, 2009, **3**, 297–303.
- 6 K. Vandewal, S. Albrecht, E. T. Hoke, K. R. Graham, J. Widmer, J. D. Douglas, M. Schubert, W. R. Mateker, J. T. Bloking, G. F. Burkhard, A. Sellinger, J. M. J. Fréchet, A. Amassian, M. K. Riede, M. D. McGehee, D. Neher and A. Salleo, *Nat. Mater.*, 2014, **13**, 63–68.
- 7 M. A. Loi, S. Toffanin, M. Muccini, M. Forster, U. Scherf and M. Scharber, *Adv. Funct. Mater.*, 2007, **17**, 2111–2116.
- 8 D. Veldman, O. Ipek, S. C. J. Meskers, J. Sweelssen, M. M. Koetse, S. C. Veenstra, J. M. Kroon, S. S. van Bavel, J. Loos and R. a. J. Janssen, *J. Am. Chem. Soc.*, 2008, **130**, 7721–7735.
- 9 D. Veldman, S. C. J. Meskers and R. A. J. Janssen, *Adv. Funct. Mater.*, 2009, **19**, 1939–1948.
- 10 M. A. Faist, T. Kirchartz, W. Gong, R. S. Ashraf, I. McCulloch, J. C. de Mello, N. J. Ekins-Daukes, D. D. C. Bradley and J. Nelson, *J. Am. Chem. Soc.*, 2012, **134**, 685–692.



- 11 H. Ohkita, S. Cook, Y. Astuti, W. Duffy, S. Tierney, W. Zhang, M. Heeney, I. McCulloch, J. Nelson, D. D. C. Bradley and J. R. Durrant, *J. Am. Chem. Soc.*, 2008, **130**, 3030–3042.
- 12 E. Wang, Z. Zhang, K. Vandewal, P. Henriksson, F. Zhang, M. R. Andersson, Z. Ma and O. Inganäs, *J. Am. Chem. Soc.*, 2011, **133**, 14244–14247.
- 13 F. Z. Koen Vandewal, Z. Ma, J. Bergqvist, Z. Tang, E. Wang, P. Henriksson, K. Tvingstedt, M. R. Andersson, O. Inganäs, O. Inganäs, K. Vandewal, Z. Ma, J. Bergqvist, Z. Tang, E. Wang, P. Henriksson, K. Tvingstedt, M. R. Andersson and F. Zhang, *Adv. Funct. Mater.*, 2012, **22**, 1–38.
- 14 H. Bronstein, E. Collado-Fregoso, A. Hadipour, Y. W. Soon, S. D. Dimitrov, R. S. Ashraf, B. P. Rand, S. E. Watkins, S. Tuladhar, I. M. Meager, J. R. Durrant, I. McCulloch, Z. Huang and P. S. Tuladhar, *Adv. Funct. Mater.*, 2013, **23**, 5647–5654.
- 15 M. Knupfer, J. Fink and D. Fichou, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2001, **63**, 1–4.
- 16 S. Holdcroft, *Macromolecules*, 1991, **24**, 4834–4838.
- 17 Y. Shimoi, A. Shuji, K. Shin-ichi and M. Kazuhiro, *Solid State Commun.*, 1995, **95**, 137–141.
- 18 B. Bernardo, D. Cheyns, B. Verreet, R. D. Schaller, B. P. Rand and N. C. Giebink, *Nat. Commun.*, 2014, **5**, 3245.
- 19 B. M. Savoie, A. Rao, A. A. Bakulin, S. Gelin, B. Movaghar, R. H. Friend, T. J. Marks and M. A. Ratner, *J. Am. Chem. Soc.*, 2014, **136**, 2876–2884.
- 20 A. A. Bakulin, A. Rao, V. G. Pavelyev, P. H. M. van Loosdrecht, M. S. Pshenichnikov, D. Niedzialek, J. Cornil, D. Beljonne and R. H. Friend, *Science*, 2012, **1340**, 1–13.
- 21 D. Amarasinghe Vithanage, a. Devizis, V. Abramavičius, Y. Infahsaeng, D. Abramavičius, R. C. I. MacKenzie, P. E. Keivanidis, a. Yartsev, D. Hertel, J. Nelson, V. Sundström and V. Gulbinas, *Nat. Commun.*, 2013, **4**, 2334.
- 22 S. Gélina, A. Rao, A. Kumar, S. L. Smith, A. W. Chin, J. Clark, T. S. van der Poll, G. C. Bazan and R. H. Friend, *Science*, 2014, **343**, 512–516.
- 23 T. Liu, D. L. Cheung and A. Troisi, *Phys. Chem. Chem. Phys.*, 2011, **13**, 21461–21470.
- 24 S. R. Yost and T. Van Voorhis, *J. Phys. Chem. C*, 2013, **117**, 5617–5625.
- 25 Y.-T. Fu, C. Risko and J.-L. Brédas, *Adv. Mater.*, 2013, **25**, 878–882.
- 26 G. F. a. Dibb, F. C. Jamieson, A. Maurano, J. Nelson and J. R. Durrant, *J. Phys. Chem. Lett.*, 2013, **4**, 803–808.
- 27 J. Knierpert and M. Schubert, *J. Phys. Chem. Lett.*, 2011, **2**, 700–705.
- 28 S. Albrecht and W. Schindler, *J. Phys. Chem. Lett.*, 2012, 640–645.
- 29 D. Caruso and A. Troisi, *Proc. Natl. Acad. Sci. U. S. A.*, 2012, **109**, 13498–13502.
- 30 E. R. Bittner and C. Silva, *Nat. Commun.*, 2014, **5**, 3119.
- 31 B. Kraabel, C. Lee, D. McBranch, D. Moses, N. Sariciftci and A. J. Heeger, *Chem. Phys. Lett.*, 1993, **213**, 389–394.
- 32 I. Hwang, D. Moses and A. Heeger, *J. Phys. Chem. C*, 2008, **112**, 4350–4354.
- 33 C.-X. Sheng, T. Basel, B. Pandit and Z. Vardeny, *Org. Electron.*, 2012, **13**, 1031–1037.
- 34 L. G. Kaake, D. Moses and A. J. Heeger, *J. Phys. Chem. Lett.*, 2013, **4**, 2264–2268.
- 35 S. Mukamel, *J. Phys. Chem. A*, 2013, **117**, 10565.
- 36 C. Brabec, G. Zerza, G. Cerullo, S. Di Silvestri, S. Luzzati, J. C. Hummelen and S. Sariciftci, *Chem. Phys. Lett.*, 2001, **340**, 232–236.
- 37 F. Zhang, K. Jespersen, C. Björström, M. Svensson, M. Andersson, V. Sundström, K. Magnusson, E. Moons, a. Yartsev and O. Inganäs, *Adv. Funct. Mater.*, 2006, **16**, 667–674.
- 38 I.-W. Hwang, C. Soci, D. Moses, Z. Zhu, D. Waller, R. Gaudiana, C. Brabec and A. J. Heeger, *Adv. Mater.*, 2007, **19**, 2307–2312.
- 39 F. Etzold, I. a. Howard, N. Forler, D. M. Cho, M. Meister, H. Mangold, J. Shu, M. R. Hansen, K. Müllen and F. Laquai, *J. Am. Chem. Soc.*, 2012, **134**, 10569–10583.
- 40 G. Grancini, M. Maiuri, D. Fazzi, A. Petrozza, H.-J. Egelhaaf, D. Brida, G. Cerullo and G. Lanzani, *Nat. Mater.*, 2013, **12**, 29–33.
- 41 N. Banerji, S. Cowan, M. Leclerc, E. Vauthey and A. J. Heeger, *J. Am. Chem. Soc.*, 2010, **132**, 17459–17470.
- 42 N. Banerji, *J. Mater. Chem. C*, 2013, **1**, 3052.
- 43 M. Tong, N. E. Coates, D. Moses, A. J. Heeger, S. Beaupré and M. Leclerc, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 125210.
- 44 F. Provencher, N. Bérubé, A. W. Parker, G. M. Greetham, M. Towrie, C. Hellmann, M. Côté, N. Stingelin, C. Silva and S. C. Hayes, *Nat. Commun.*, 2014, **5**, 4288.
- 45 L. Onsager, *J. Chem. Phys.*, 1934, **2**, 599.
- 46 C. L. Braun, *J. Chem. Phys.*, 1984, **80**, 4157.
- 47 V. Mihailetschi, L. Koster, J. Hummelen and P. Blom, *Phys. Rev. Lett.*, 2004, **93**, 19–22.
- 48 P. Peumans and S. R. Forrest, *Chem. Phys. Lett.*, 2004, **398**, 27–31.
- 49 C. Deibel, T. Strobel and V. Dyakonov, *Phys. Rev. Lett.*, 2009, **103**, 1–4.
- 50 V. I. Arkhipov, P. Heremans and H. Baißler, *Appl. Phys. Lett.*, 2003, **82**, 4605.
- 51 A. V. Nenashev, S. D. Baranovskii, M. Wiemer, F. Jansson, R. Österbacka, a. V. Dvurechenskii and F. Gebhard, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 035210.
- 52 T. Offermans, S. C. J. Meskers and R. a. J. Janssen, *Chem. Phys.*, 2005, **308**, 125–133.
- 53 H. van Eersel, R. a. J. Janssen and M. Kemerink, *Adv. Funct. Mater.*, 2012, **22**, 2700–2708.
- 54 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 1372.
- 55 G. Zhang and C. B. Musgrave, *J. Phys. Chem. A*, 2007, **111**, 1554–1561.
- 56 A. Dreuw and M. Head-Gordon, *J. Am. Chem. Soc.*, 2004, **126**, 4007–4016.
- 57 D. J. Tozer, *J. Chem. Phys.*, 2003, **119**, 12697.
- 58 T. Ziegler, M. Seth, M. Krykunov, J. Autschbach and F. Wang, *THEOCHEM*, 2009, **914**, 106–109.
- 59 P. M. W. Gill, R. D. Adamson and J. a. Pople, *Mol. Phys.*, 1996, **88**, 1005–1009.



- 60 T. Leininger, H. Stoll, H.-J. Werner and A. Savin, *Chem. Phys. Lett.*, 1997, **275**, 151–160.
- 61 H. Iikura, T. Tsuneda, T. Yanai and K. Hirao, *J. Chem. Phys.*, 2001, **115**, 3540.
- 62 T. Yanai, D. P. Tew and N. C. Handy, *Chem. Phys. Lett.*, 2004, **393**, 51–57.
- 63 M. J. G. Peach, P. Benfield, T. Helgaker and D. J. Tozer, *J. Chem. Phys.*, 2008, **128**, 044118.
- 64 R. Baer and D. Neuhauser, *Phys. Rev. Lett.*, 2005, **94**, 2–5.
- 65 E. Livshits and R. Baer, *Phys. Chem. Chem. Phys.*, 2007, **9**, 2932–2941.
- 66 L. Hedin, *Phys. Rev.*, 1965, **139**, 796–823.
- 67 E. Salpeter and H. Bethe, *Phys. Rev.*, 1951, **84**, 1232–1242.
- 68 Y. Ma, M. Rohlfing and C. Molteni, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2009, **80**, 241405.
- 69 B. Baumeier, D. Andrienko and M. Rohlfing, *J. Chem. Theory Comput.*, 2012, **8**, 2790–2795.
- 70 B. Baumeier, D. Andrienko, Y. Ma and M. Rohlfing, *J. Chem. Theory Comput.*, 2012, **8**, 997–1002.
- 71 D. Niedzialek, I. Duchemin, T. B. D. Queiroz, S. Osella, A. Rao, R. Friend, X. Blase, S. Kümmel and D. Beljonne, 2014, DOI: 10.1002/adfm.201402682.
- 72 E. B. Isaacs, S. Sharifzadeh, B. Ma and J. B. Neaton, *J. Phys. Chem. Lett.*, 2011, **2**, 2531–2537.
- 73 Y.-S. Huang, S. Westenhoff, I. Avilov, P. Sreearunothai, J. M. Hodgkiss, C. Deleener, R. H. Friend and D. Beljonne, *Nat. Mater.*, 2008, **7**, 483–489.
- 74 S. Difley and T. V. Voorhis, *J. Chem. Theory Comput.*, 2011, **7**, 594–601.
- 75 K. E. Riley, B. T. Op't Holt and K. M. Merz, *J. Chem. Theory Comput.*, 2007, **3**, 407–433.
- 76 M. Swart, L. Jensen and P. Van Duijnen, *Solvation Effects on Molecules and Biomolecules*, Springer, Netherlands, Dordrecht, 2008, pp. 39–102.
- 77 S. Difley, L.-P. Wang, S. Yeganeh, S. R. Yost and T. Van Voorhis, *Acc. Chem. Res.*, 2010, **43**, 995–1004.
- 78 Y. Kanai and J. C. Grossman, *Nano Lett.*, 2007, **7**, 1967–1972.
- 79 I. H. Nayyar, E. R. Batista, S. Tretiak, A. Saxena, D. L. Smith and R. L. Martin, *J. Phys. Chem. Lett.*, 2011, **2**, 566–571.
- 80 R. M. Martin, *Electronic structure: basic theory and practical methods*, Cambridge University Press, 2004.
- 81 D. Beljonne, J. Cornil, L. Muccioli, C. Zannoni, J.-L. Breidas and F. Castet, *Chem. Mater.*, 2011, **23**, 591–609.
- 82 S. Few, J. M. Frost, J. Kirkpatrick and J. Nelson, *J. Phys. Chem. C*, 2014, **16**, 8253–8261.
- 83 M. a. Loi, S. Toffanin, M. Muccini, M. Forster, U. Scherf and M. Scharber, *Adv. Funct. Mater.*, 2007, **17**, 2111–2116.
- 84 D. Veldman, O. Ipek, S. C. J. Meskers, J. Sweelssen, M. M. Koetse, S. C. Veenstra, J. M. Kroon, S. S. van Bavel, J. Loos and R. A. J. Janssen, *J. Am. Chem. Soc.*, 2008, **130**, 7721–7735.
- 85 A. A. Y. Guilbert, M. Schmidt, A. Bruno, J. Yao, S. King, M. Sachetan, T. Kirchartz, M. I. Alonso, A. R. Goñi, N. Stingelin, A. Saif, M. Campoy-quiles and J. Nelson, *Adv. Funct. Mater.*, 2014, **24**, 6972–6980.
- 86 B. A. C. Mayer, M. F. Toney, S. R. Scully, J. Rivnay, C. J. Brabec, M. Scharber, M. Koppe, M. Heeney, I. Mcculloch and M. D. Mcgehee, *Adv. Funct. Mater.*, 2009, 1173–1179.
- 87 W. L. Rance, A. J. Ferguson, T. McCarthy-Ward, M. Heeney, D. S. Ginley, D. C. Olson, G. Rumbles and N. Kopidakis, *ACS Nano*, 2011, **5**, 5635–5646.
- 88 D. L. Cheung and A. Troisi, *J. Phys. Chem. C*, 2010, **114**, 20479–20488.
- 89 M. T. Rispens, A. Meetsma, R. Rittberger, C. J. Brabec, N. S. Sariciftci and J. C. Hummelen, *Chem. Commun.*, 2003, 2116–2118.
- 90 J. J. Kwiatkowski, J. M. Frost and J. Nelson, *Nano Lett.*, 2009, **9**, 1085–1090.
- 91 G. Raos, M. Casalegno and J. Ide, *J. Chem. Theory Comput.*, 2013, **10**, 364–372.
- 92 I. Dabo, A. Ferretti, C.-H. Park, N. Poilvert, Y. Li, M. Cococcioni and N. Marzari, *Phys. Chem. Chem. Phys.*, 2013, **15**, 685–695.
- 93 S. Verlaak, D. Beljonne, D. Cheyns, C. Rolin, M. Linares, F. Castet, J. Cornil and P. Heremans, *Adv. Funct. Mater.*, 2009, **19**, 3809–3814.
- 94 P. C. Eklund, A. M. Rae, Y. Wang, P. Zhou, K.-a. Wang, J. M. Holden, M. S. Dresselhaus and G. Dresselhaus, *Thin Solid Films*, 1995, **257**, 211.
- 95 M. Alonso, M. Garriga, N. Karl, J. Ossó and F. Schreiber, *Org. Electron.*, 2002, **3**, 23–31.
- 96 M. Campoy-Quiles, M. I. Alonso, D. D. C. Bradley and L. J. Richter, *Adv. Funct. Mater.*, 2014, **24**, 2116–2134.
- 97 M. Campoy-Quiles, C. Müller, M. Garriga, E. Wang, O. Inganäs and M. Alonso, *Thin Solid Films*, 2014, DOI: 10.1016/j.tsf.2014.02.096.
- 98 H. Ishii, K. Sugiyama, E. Ito and K. Seki, *Adv. Mater.*, 1999, **11**, 605–625.
- 99 S. C. Veenstra and H. T. Jonkman, *J. Polym. Sci., Part B: Polym. Phys.*, 2003, **41**, 2549–2560.
- 100 O. Molodtsova, T. Schwieger and M. Knupfer, *Appl. Surf. Sci.*, 2005, **252**, 143–147.
- 101 Y. Ge and J. E. Whitten, *Chem. Phys. Lett.*, 2007, **448**, 65–69.
- 102 W. Osikowicz, M. deJong and W. Salaneck, *Adv. Mater.*, 2007, **19**, 4213–4217.
- 103 P. B. Allen, *J. Chem. Phys.*, 2004, **120**, 2951.
- 104 J. Applequist, J. R. Carl and K.-k. Fung, *J. Am. Chem. Soc.*, 1972, **261**, 2952–2960.
- 105 B. Thole, *Chem. Phys.*, 1981, **59**, 341–350.
- 106 P. P. T. van Duijnen and M. Swart, *J. Phys. Chem. A*, 1998, **102**, 2399–2407.
- 107 H. a. Stern, G. a. Kaminski, J. L. Banks, R. Zhou, B. J. Berne and R. A. Friesner, *J. Phys. Chem. B*, 1999, **103**, 4730–4737.
- 108 E. Tsiper and Z. Soos, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **68**, 085301.
- 109 M. Swart and P. van Duijnen, *Mol. Simul.*, 2006, **32**, 471–484.
- 110 V. Lacivita, M. Rerat, R. Orlando, M. Ferrero and R. Dovesi, *J. Chem. Phys.*, 2012, **136**, 114101.
- 111 M. van Faassen, P. de Boeij, R. van Leeuwen, J. Berger and J. Snijders, *Phys. Rev. Lett.*, 2002, **88**, 186401.



- 112 D. S. Sabirov, R. R. Garipova and R. G. Bulgakov, *Chem. Phys. Lett.*, 2012, **523**, 92–97.
- 113 K. Kowalski, J. R. Hammond, W. a. de Jong and A. J. Sadlej, *J. Chem. Phys.*, 2008, **129**, 226101.
- 114 M. Huzak and M. S. Deleuze, *J. Chem. Phys.*, 2013, **138**, 024319.
- 115 M. Linares, D. Beljonne, K. Lancaster, J.-l. Bre, S. Verlaak, A. Mityashin, P. Heremans, A. Fuchs, C. Lennartz, P. Aurel and L. Ducasse, *J. Phys. Chem. C*, 2010, 3215–3224.
- 116 C. Chia, D. Teo, S. Few and J. Nelson, 2014, manuscript in preparation.
- 117 C. Poelking, M. Tietze, C. Elschner, S. Olthof, D. Hertel, W. Frank, K. Meerholz, K. Leo and D. Andrienko, under review.
- 118 H. D. de Gier, R. Broer and R. W. Havenith, *Phys. Chem. Chem. Phys.*, 2014, **16**, 12454–12461.
- 119 Y. Yi, V. Coropceanu and J.-L. Brédas, *J. Mater. Chem.*, 2011, **21**, 1479.
- 120 T. Liu and A. Troisi, *J. Phys. Chem. C*, 2011, **115**, 2406–2415.
- 121 H. Tamura, I. Burghardt and M. Tsukada, *J. Phys. Chem. C*, 2011, **115**, 10205–10210.
- 122 H. Tamura and I. Burghardt, *J. Am. Chem. Soc.*, 2013, **135**, 16364–16367.
- 123 C. Moser, J. Keske, K. Warncke, R. Farid and P. Dutton, *Nature*, 1992, **355**, 796–802.
- 124 J. N. Clifford, E. Palomares, M. K. Nazeeruddin, M. Grätzel, J. Nelson, X. Li, N. J. Long and J. R. Durrant, *J. Am. Chem. Soc.*, 2004, **126**, 5225–5233.
- 125 B. Albinsson and J. Martensson, *J. Photochem. Photobiol., C*, 2008, **9**, 138–155.

