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# Quantum size effect and catalytic activity of nanosized single-crystalline spherical $\beta$ -Ga<sub>2</sub>O<sub>3</sub> particles by thermal annealing of liquid metal nanoparticles†

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We report on the synthesis of nanosized single-crystalline spherical  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> particles (GONP) by thermal annealing of gallium nanoparticles (GaNP), which previously has been prepared under ultrasonication. GONP displayed a spherical morphology and the particle size reflected that of GaNP. The nanoparticles have been confirmed as crystalline  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> by powder X-ray diffraction. The selected area electron diffraction patterns obtained from a whole particle of one GONP represented production of single-crystalline nanoparticle. Furthermore, the dispersibility and stability of GONP in solvent dramatically improved through silica coating (GO@SiO<sub>2</sub> NP). Photoluminescence spectroscopy indicated that the GO@SiO<sub>2</sub> NP exhibited a light blue emission at 445 nm under excitation at 240 nm. Diffuse reflectance spectroscopy demonstrated that the band gap of GO@SiO<sub>2</sub> NP ( $E_g = 4.89$  eV) was hypsochromically shifted compared to that of bulk Ga<sub>2</sub>O<sub>3</sub> ( $E_g(\text{bulk}) = 4.56$  eV) due to the quantum size effect of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>. Additionally, silica particle-supported GONP (GONP/MCM-41) showed catalytic activity for the benzylation of benzene with benzyl chloride. Thus, the GONP presented here will have potential applications in LED phosphors, optoelectronic devices, and supported catalysts.

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## Introduction

Monoclinic gallium oxide ( $\beta$ -Ga<sub>2</sub>O<sub>3</sub>) is a well-known wide-band-gap semiconductor ( $E_g > 4.4$  eV) that exhibits luminescence and conduction properties.<sup>1–4</sup> Due to these unique performances,  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> can be used for a variety of applications including electronic and optoelectronic devices, such as high sensitivity gas sensors, and transparent conductors.<sup>5–10</sup> Up to now, micrometer/nanometer scaled single-crystalline  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> with varied morphology, for instance, rods, wires, tubes, plates have been synthesized and characterized.<sup>11–19</sup> However, examples of the preparation of single-crystalline spherical  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> particles has still remained very few. Recently, Tang *et al.* have reported a simple method to prepare submicrometer-sized spherical  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> particles with uniform diameters.<sup>20</sup> The  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> submicron particle was synthesized by heating molten gallium, and the resulting particles exhibited a broad emission spectrum with a maximum at 458 nm under 260 nm excitation, since various quantum size effects in electronic, optical, and/or magnetic properties, such as an increase in the band gap of a semiconducting material with a reduction of the size, are

generally observed for other materials.<sup>21–27</sup> Downsizing of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> submicron particle will also provide distinctive quantum size effect on these properties. Nevertheless, to the best of our knowledge, preparation of single-crystalline spherical  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> nanoparticles and the quantum size effect have been never reported yet.

These facts prompted us to explore the quantum size effect of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> particles. We previously reported preparation of liquid-metal nanocolloid composed of liquid gallium, solvents, and surfactants with ultrasonic irradiation.<sup>28</sup> The direct synthesis method enabled to control the size of gallium nanoparticle (GaNP) by tuning the surfactants, temperature, and ultrasonic irradiation intensity.

In this report, we demonstrate for the first time preparation of single-crystalline spherical  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> nanoparticle (GONP) derived from GaNP as precursor (Fig. 1). To suppress the aggregation of prepared GONP, we selected silica coated GaNP (Ga@SiO<sub>2</sub> NP) as precursors of GONP. The obtained GONP exhibited luminescence property and hypsochromic shift of band gap due to quantum size effect of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>. Furthermore, the GONP were supported on MCM-41, resulting in catalytic activity for Friedel–Crafts type benzylation.

## Experimental section

### Materials and instrumentation

Gallium (high purity, 99.9999%, Mitsuwa Chemicals Co., Ltd.), dodecylamine (Tokyo Chemical Industry Co., Ltd.), oleic acid

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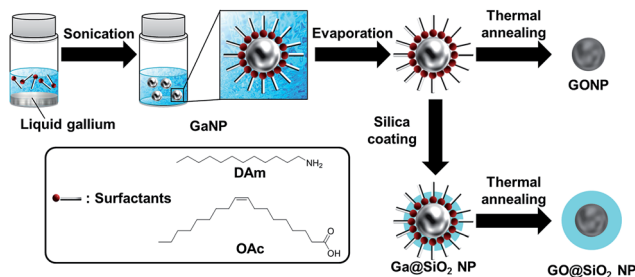


Fig. 1 Schematic illustration of preparation of spherical  $\beta$ - $\text{Ga}_2\text{O}_3$  nanoparticle (GONP) and silica-coated  $\beta$ - $\text{Ga}_2\text{O}_3$  nanoparticle ( $\text{GO@SiO}_2$  NP) from gallium nanoparticle (GaNP) as precursors. Surfactants are dodecylamine (DAm) or oleic acid (OAc).

(Wako chemical industries, Ltd.), tetramethyl orthosilicate (Tokyo Chemical Industry Co., Ltd.), chloroform (Wako chemical industries, Ltd.), 2-propanol (Wako chemical industries, Ltd.) were purchased from each company, and used without further purification. Sonication was carried out with ELMA Elmasonic S30H and SND US-2KS (80 W for  $\text{GaNPs@DAm}$  and 120 W for  $\text{GaNPs@OAc}$ ). Powder X-ray diffraction (XRD) patterns were obtained by Bruker AXS D8 ADVANCE. Transmission electron microscopy (TEM) was conducted on a JEOL JEM-2100F having EDX function and a JEOL JEM-2010 for selected-area electron diffraction (SAED). Emission spectra were obtained with SHIMADZU RF5300PC spectrofluorometer. UV-Vis spectra were recorded on a JASCO V-570 spectrophotometer.  $^1\text{H}$  (500 MHz) NMR spectra were measured on a Bruker Biospin AVANCE DRX500 instrument, using 0.05% tetramethylsilane (TMS) as an internal standard. TEM observation was conducted at the OPEN FACILITY, Hokkaido University Sousei Hall.

### Preparation of GaNP suspension<sup>28</sup>

Metallic gallium (100 mg, 1.4 mmol) was added to a chloroform solution (2.4 mL) of dodecylamine (276  $\mu\text{L}$ ), and the solution was heated at 80  $^\circ\text{C}$  for 15 min. After visual confirmation of gallium melting, the solution was sonicated by a bath type sonicator at 0  $^\circ\text{C}$  for 10 min, which resulted in dodecylamine stabilized GaNP ( $\text{GaNP@DAm}$ ) suspension.

### Preparation of GONP

$\text{GaNP@DAm}$  suspension was separated by centrifugation, and the  $\text{GaNP@DAm}$  powder was obtained by drying. GONP was prepared by thermal annealing under air using electric furnace and oven at 500–900  $^\circ\text{C}$  under ambient atmosphere for 1 hour.

### Preparation of $\text{SiO}_2$ coated GaNP ( $\text{Ga@SiO}_2$ NP)

Metallic gallium (100 mg, 1.4 mmol) was added to a chloroform solution (2.4 mL) of oleic acid (381  $\mu\text{L}$ ), and the solution was heated at 80  $^\circ\text{C}$  for 15 min. After confirmation of melting gallium, the solution was sonicated by bath type sonicator at 0  $^\circ\text{C}$  for 10 min, which resulted in oleic acid stabilized GaNP ( $\text{GaNP@OAc}$ ) suspension.  $\text{GaNP@OAc}$  suspension was separated by centrifugation, and the  $\text{GaNP@OAc}$  powder was

obtained by drying. Then, the  $\text{GaNP@OAc}$  powder was added to 90 mL 2-propanol, 7.2 mL water, and 900  $\mu\text{L}$  ammonia solution. Under continuous stirring, 0.9 mg of tetramethyl orthosilicate (TMOS) was added to the reaction mixture. The reaction was allowed to proceed for 1 hour at room temperature under continuous stirring, which resulted in silica coated  $\text{GaNP@OAc}$  ( $\text{Ga@SiO}_2$  NP) suspension.

### Preparation of silica coated GONP ( $\text{GO@SiO}_2$ NP)

$\text{Ga@SiO}_2$  NP suspension was separated by centrifugation and  $\text{Ga@SiO}_2$  NP powder were obtained by drying. Silica coated GONP ( $\text{GO@SiO}_2$  NP) was prepared by thermal annealing using electric furnace and oven at 700  $^\circ\text{C}$  for 1 hour.

### Preparation of MCM-41 supported GONP ( $\text{GONP/MCM-41}$ )

GONP supported on MCM-41 ( $\text{GONP/MCM-41}$ ) was synthesized by evaporation to dryness. MCM-41 (20 mg) was dispersed in chloroform (6 mL). Then,  $\text{GaNP@OAc}$  powder (2 mg) was added to MCM-41 dispersion. After ultrasonication, the mixture was evaporated to dryness, which resulted in  $\text{GaNP@OAc}$  supported on MCM-41. Supported GONP was prepared from  $\text{GaNP@OAc}$  supported on MCM-41 by thermal annealing using electric furnace and oven at 700  $^\circ\text{C}$  for 1 hour.

### Evaluation for catalytic activity of supported GONP

The liquid phase benzylation of benzene with benzyl chloride ( $\text{BzCl}$ ) was selected as a probe reaction for studying the activity of catalysts. The reaction was carried out in a 25 mL round bottomed flask under continuous stirring equipped with condenser, a nitrogen inlet for maintaining an inert atmosphere. The temperature was maintained in oil bath and the reaction carried out at selected reaction condition with reaction temperature of 353 K, benzene to  $\text{BzCl}$  molar ratio of 10, and the catalyst amount of 1 wt% of the total reaction mixture. After 3 hours, the sample was withdrawn and analyzed with  $^1\text{H}$  NMR measurement. The conversion was based on the consumed in the reaction mixture.

## Results and discussion

The GONP powder was obtained from  $\text{GaNP@DAm}$  after annealing under air for 1 hour at various temperatures. In the case of annealing temperature above 700  $^\circ\text{C}$ , the obtained powder was colourless, while that annealed below 600  $^\circ\text{C}$  exhibited gray color. Their crystal phase was identified by XRD, as shown Fig. 2. Whereas the XRD patterns of GONP powders treated above 700  $^\circ\text{C}$  confirms that most of the strong peaks corresponded with commercial  $\beta$ - $\text{Ga}_2\text{O}_3$ , powders treated below 600  $^\circ\text{C}$  reveal low crystalline nature. This fact indicated the importance of annealing temperature on the crystallinity of GONP.

The morphology of products obtained after thermal annealing with various temperatures was investigated by transmission electron microscope (TEM) as shown Fig. 3. It clearly shows nanosized spherical particles treated below 700  $^\circ\text{C}$  (Fig. 3b–d), derived from the original  $\text{GaNP@DAm}$  particle



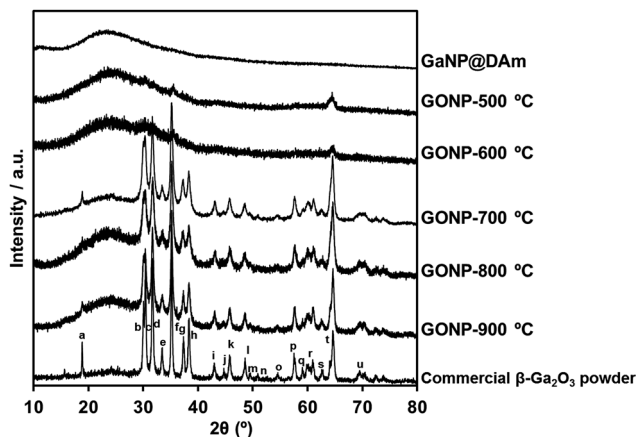


Fig. 2 XRD patterns of the powders annealed with different temperatures of GaNP@DAm before annealing, GONP-500 °C, GONP-600 °C, GONP-700 °C, GONP-800 °C, GONP-900 °C, and commercial  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> powder. a: (−201), b: (400), c: (−401), d: (002), (−202), e: (−111), f: (111), g: (401), h: (−311), (−402), (202), i: (−112), j: (−601), k: (600), (−312), (112), l: (510), m: (402), n: (−403), o: (511), (203), p: (−313), q: (113), r: (−801), s: (710), t: (403), (512), (−222), u: (−421), (022), (−222).

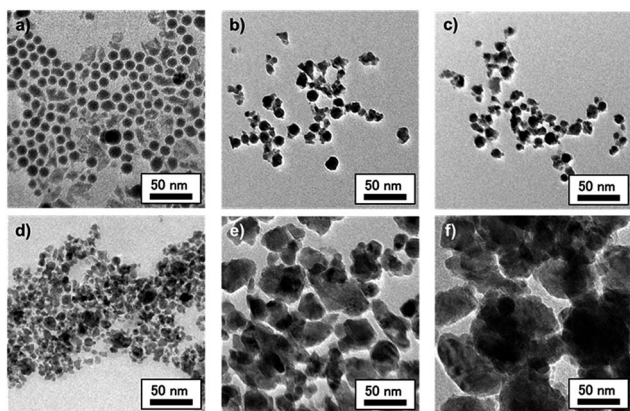


Fig. 3 TEM images of (a) GaNP@DAm before annealing, and GONP annealed at (b) 500 °C, (c) 600 °C, (d) 700 °C, (e) 800 °C, and (f) 900 °C.

(Fig. 3a). Indeed, the measured particle distribution of GONP annealed at 700 °C ( $15.9 \pm 4.2$  nm) was similar with that of GaNP@DAm ( $13.3 \pm 4.9$  nm, Fig. S1†), while the GONP partially aggregated to form larger particle (Fig. 3d). With the increase of thermal annealing temperature, the particle was likely to aggregate and formed bulky crystal with irregular morphologies, probably due to higher oxidation reaction rate (Fig. 3e and f).

High magnification TEM images of a single particle annealed at various temperatures are shown in Fig. 4a–e. The crystalline nature of these particles was verified by the high-resolution TEM (HRTEM), as shown in Fig. 4f–j. The marked interplanar  $d$ -spacing of all samples corresponded with facet of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> such as (403), (400) or (111). However for particles treated below 600 °C, the mixed population of crystalline part and amorphous part in a single particle was observed (Fig. 4f

and g). Thus, crystallization of GONP was completely proceeded above 700 °C.<sup>29</sup>

In order to confirm its single-crystalline character of the particles treated above 700 °C, we obtained selected-area electron diffraction (SAED) pattern from a single particle (Fig. 5). From the SAED patterns of single particle annealed at 700 °C, all the diffraction spots can be indexed to single-crystalline  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (Fig. 5a). On the other hand, the SAED patterns of single particle annealed at 900 °C revealed poly crystalline  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> as can be seen in the unindexed spots marked by yellow circle in Fig. 5b.

The result of preparation of GONP was summarized in Table 1. These observations offered two facts for preparation of GONP. At first, when thermal annealing temperature was 500 °C or 600 °C, crystallization of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> was not completely proceeded with the result that gallium and  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> mixed nanoparticles was formed. In the case of thermal annealing temperature above 700 °C, gallium part of GaNP@DAm as precursors perfectly transformed into  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>, resulted in formation of single-crystalline GONP at 700 °C. However for the second fact, aggregation and fusion of single-crystalline GONP was occurred at 800 °C or 900 °C, which brought about the formation of polycrystalline GONP.

Then we focused characteristic and property of single-crystalline GONP annealed at 700 °C. We investigated the emission property of the GONP. The absorption, excitation, and emission spectra of the GONP dispersed in methanol were measured at room temperature, as shown in Fig. 6. The emission spectrum shows a broad light blue emission band centered at 451 nm under a 248 nm excitation. Note that GONP prepared at 900 °C exhibited the emission maximum at 470 nm, which was identical to that of commercial  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> powder, indicative of the importance of single crystalline nature for the emission property. The band gap ( $E_g$ ) of GONP was estimated by using a Tauc plot of UV-Vis diffuse reflectance spectroscopy (Fig. 6b), which was found to be 4.84 eV. The commercial  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> powder showed its band gap at 4.56 eV, thus a quantum size effect of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> caused the 0.28 eV of hypsochromic shift on the band gap, corresponding to the anticipated value from Brus equation (Fig. S2†).<sup>30,31</sup>

The obtained single crystalline GONP slightly aggregated due to disappearance of stabilizer through thermal annealing, as shown in Fig. 3d. To suppress aggregation of GONP by coating GaNP with heat-resistant materials, we attempted to prepare silica coated GONP (GO@SiO<sub>2</sub> NP) utilizing silica coated GaNP (Ga@SiO<sub>2</sub> NP) as precursors to suppress aggregation of nanoparticles. Moreover, we selected smaller GaNP (GaNP@OAc; average diameter: *ca.* 8 nm) compared to GaNP@DAm (average diameter: *ca.* 15 nm) as the precursor. Ga@SiO<sub>2</sub> NP was obtained through a sol-gel reaction of tetramethyl orthosilicate (TMOS) catalyzed by aqueous ammonia as a catalyst in alcoholic solvent at room temperature. An excess amount of TMOS was employed due to the instability of GaNP in an alcoholic solution. After the sol-gel reaction for 1 hour, GO@SiO<sub>2</sub> NP was prepared by thermal annealing at 700 °C for 1 hour. Thermal annealing caused colour change of powder from black to white. Whereas the XRD patterns of





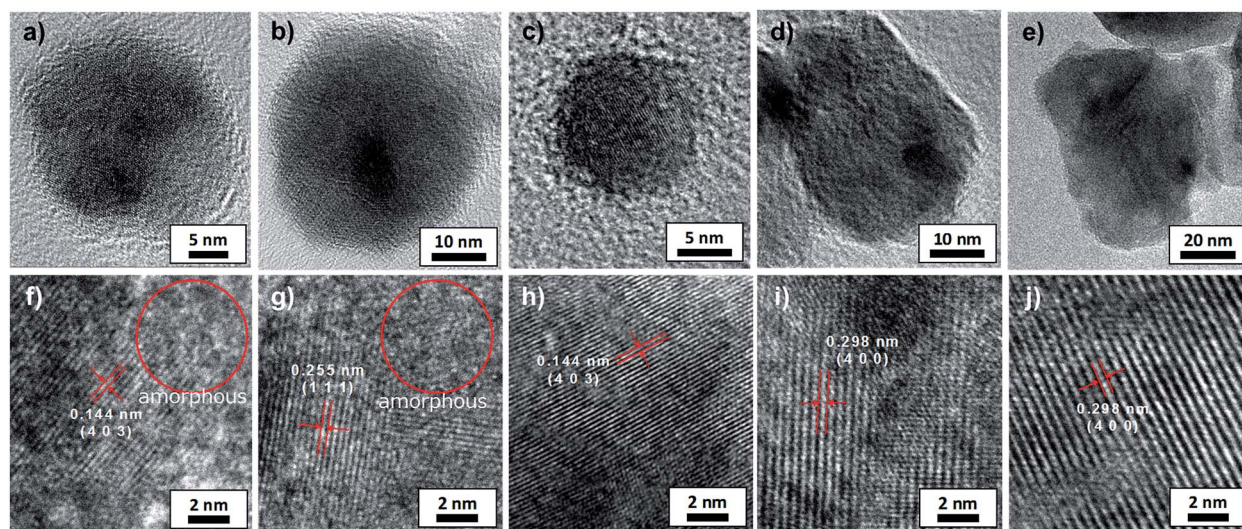


Fig. 4 TEM images of single particle of GONP annealed at different temperatures: (a) 500 °C, (b) 600 °C, (c) 700 °C, (d) 800 °C, and (e) 900 °C. HRTEM images of single particle prepared at different temperatures: (f) 500 °C, (g) 600 °C, (h) 700 °C, (i) 800 °C, and (j) 900 °C.

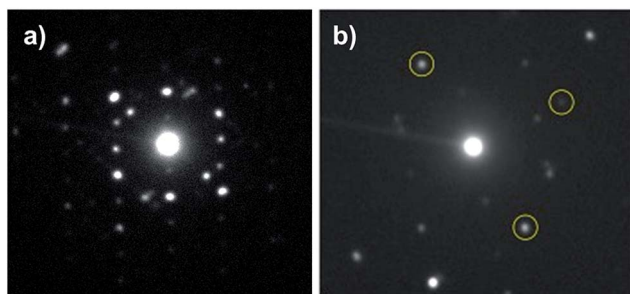


Fig. 5 SAED patterns of single particle of GONP annealed at (a) 700 °C, and (b) 900 °C.

**Ga@SiO<sub>2</sub> NP** powder shows amorphous, powders treated at 700 °C confirms weak peaks at 35.3° and 64.3°, respectively attributed to (111) and (403) of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (Fig. 7).

TEM observation of resulted particles revealed the formation of **GO@SiO<sub>2</sub> NP**, which showed similar appearance with **Ga@SiO<sub>2</sub> NP** (Fig. 8a and b). The distribution of particle diameter in **GO@SiO<sub>2</sub> NP** ( $9.1 \pm 1.6$  nm) was close to that of **Ga@SiO<sub>2</sub> NP** ( $9.6 \pm 1.8$  nm, Fig. S3†). EDX-analysis of **GO@SiO<sub>2</sub> NP** proved the existence of Ga and Si atoms as well as those in

**Ga@SiO<sub>2</sub> NP**. Then we investigated the emission property and band gap of **GO@SiO<sub>2</sub> NP**. Analogous to **GONP**, emission spectrum of **GO@SiO<sub>2</sub> NP** showed a broad emission spectrum centered at 446 nm under a 240 nm excitation, which was caused hypsochromic shift compared with **GONP** (Fig. S4†). Furthermore, the band gap of **GO@SiO<sub>2</sub> NP** was determined as 4.89 eV from Tauc plot, thus larger than that of **GONP** (Fig. 6b). These results confirmed existence of quantum size effect of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>.

To investigate the catalytic activity of **GONP**, benzylation of benzene with benzyl chloride (**BzCl**) has been carried out.<sup>32–34</sup> **GONP/MCM-41** was used for benzylation at the reaction temperature of 80 °C, the reaction time of 3 hours and the benzene to **BzCl** ratio 10 and the results are shown Fig. 9. The major product in the reaction is diphenylmethane (**DPM**). The trace of the formation of by-product (dibenzylbenzene) was found. **BzCl** conversion and **DPM** selectivity were calculated to be 71% and 51%, respectively. The time of frequency (TOF) of this catalysis was found to be 25 h<sup>-1</sup>. On the other hand, commercial  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> powder did not show formation of **DPM** at all. This result revealed that our catalyst was active for the benzylation of benzene with **BzCl**, probably due to the increased surface area of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>.

Table 1 The characteristics of GaNP@DAm and GONP prepared under various condition

Sample	Color	Shape	Composition	Crystallinity
GaNP@DAm	Black	Sphere	Amorphous	—
GONP-500 °C	Gray	Sphere	Amorphous + crystal	—
GONP-600 °C	Gray	Sphere	Amorphous + crystal	—
GONP-700 °C	White	Sphere	Crystal	Single crystal
GONP-800 °C	White	Non-sphere	Crystal	Poly crystal
GONP-900 °C	White	Non-sphere	Crystal	Poly crystal



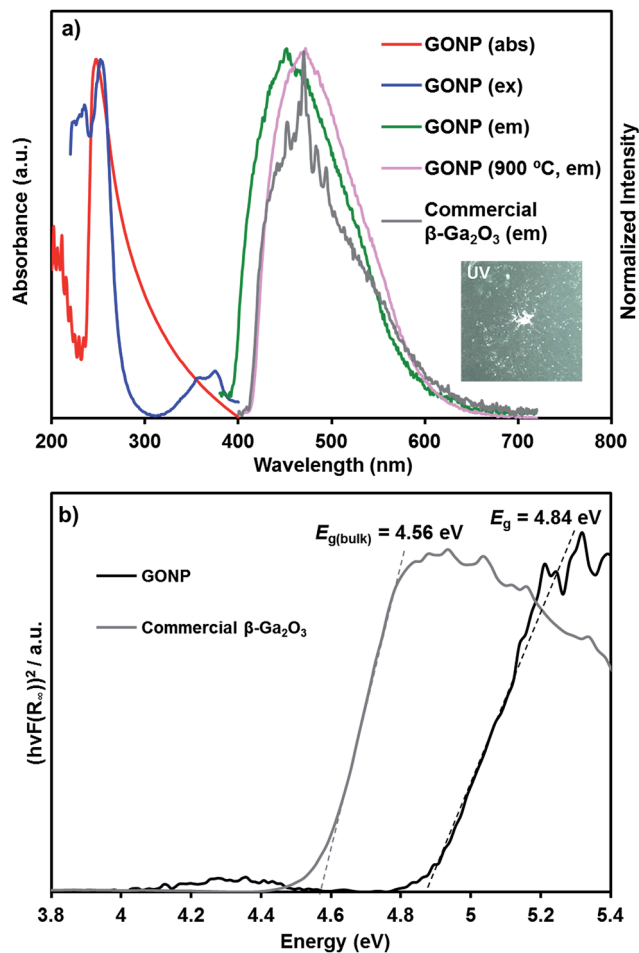


Fig. 6 (a) Absorption, excitation ( $\lambda_{\text{em}} = 451 \text{ nm}$ ) and emission spectra ( $\lambda_{\text{ex}} = 248 \text{ nm}$ ) of single-crystalline GONP and commercial  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> powder. Emission spectrum of GONP prepared at 900 °C was also shown. Inset shows the photograph of single-crystalline GONP under UV irradiation at 254 nm. (b) Band gap data of single-crystalline GONP and commercial  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> powder obtained from UV reflectance.

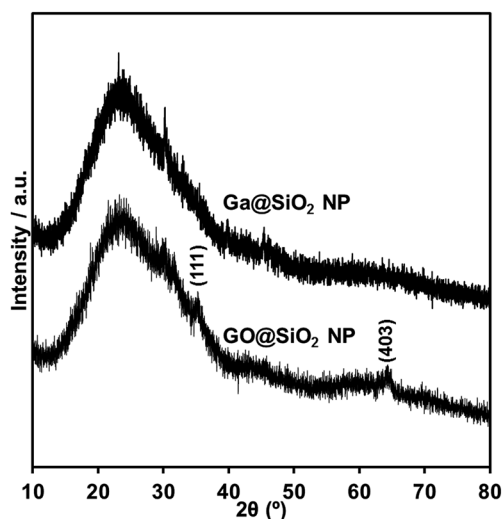


Fig. 7 XRD patterns of silica coated GaNP@OAc (Ga@SiO<sub>2</sub> NP) and silica coated GONP (GO@SiO<sub>2</sub> NP).

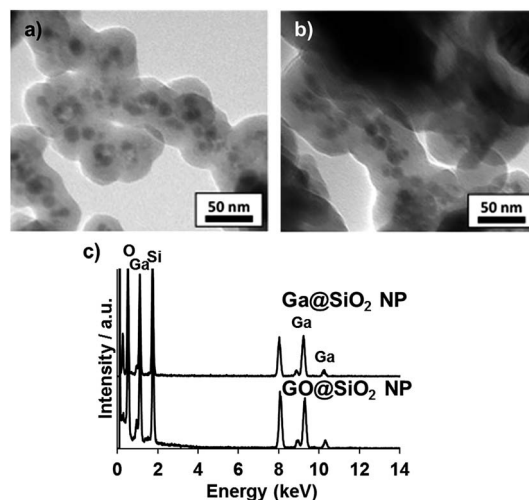


Fig. 8 TEM images of (a) Ga@SiO<sub>2</sub> NP, and (b) GO@SiO<sub>2</sub> NP, and (c) TEM-EDX analysis of Ga@SiO<sub>2</sub> NP and GO@SiO<sub>2</sub> NP.

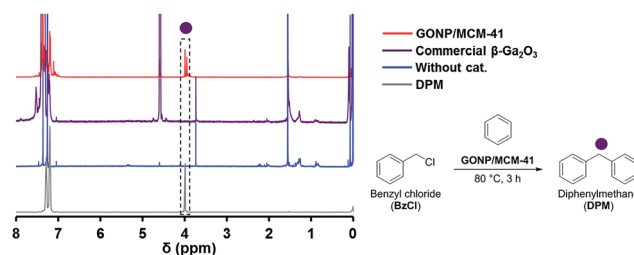


Fig. 9 <sup>1</sup>H NMR spectra of sample after benzylation under various conditions.

## Conclusions

We have demonstrated a novel approach to prepare nanosized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> particles utilizing GaNP as precursors. A strong dependence between the structure of GONP and the thermal annealing temperature was observed. Furthermore, it was found that silica coating allowed to maintain the spherical shape of nanoparticles without formation of aggregation and change of morphology (GO@SiO<sub>2</sub> NP). The single-crystalline GONP with spherical morphology exhibited a light blue emission and band gap shift derived from quantum size effect compared to bulk  $\beta$ -Ga<sub>2</sub>O<sub>3</sub>. Moreover, MCM-41 supported GONP (GONP/MCM-41) revealed catalytic activity for benzylation of benzene with BzCl. We strongly believe that GONP will be suitable for applications in LED phosphors, optoelectronic devices, and supported catalysts.

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observation was carried out at the OPEN FACILITY, Hokkaido University Sousei Hall.

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