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Ultra-narrow blue phosphorene nanoribbons for tunable optoelectronics†

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We report optoelectronic properties of ultra-narrow blue phosphorene nanoribbons (BPNRs) within the state-of-the-art density functional theory framework. The positive but small value of formation energy (~ 0.1 eV per atom) indicates the relative ease of the formation of BPNRs from their two-dimensional (2D) counterpart. The oscillatory behaviour of the electronic band gap of bare BPNRs with increasing width is attributed to the reconstruction of edge atoms. The static dielectric constant of BPNRs depends on the width and applied strain which in turn shows consistency with the Penn's model expression for semiconductors. Bare BPNRs exhibit both π and $\pi + \sigma$ plasmonic structures while passivated ones possess only a $\pi + \sigma$ plasmonic structure that get blue-shifted (as large as ~ 3 eV) on increasing the width of the BPNRs which makes electron energy loss spectroscopy useful for identifying the width of BPNRs in real experimental situations. The mechanical strain induces a small red shift in, which is attributed to the modification in electronic band dispersion due to a different superposition of atomic orbitals on the application of applied strain. These tunable electronic and dielectric properties of BPNRs mean they may find applications in optoelectronic devices based on blue phosphorene.

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1. Introduction

After the discovery of graphene in 2004, it became the world's first two dimensional (2D) stable crystal with a honeycomb structure.¹ Graphene possess unique electronic and mechanical properties along with high carrier mobility, even greater than Si at room temperature.²⁻⁴ Graphene gave rise to a new era of low dimensional materials, consequently, new classes of 2D materials such as silicene,^{5,6} germanene,⁷ transition metal dichalcogenide (TMDs)⁸⁻¹⁰ *etc.* were predicted and have been realised experimentally. TMDs exist in variety of forms ranging from metals to wide-gap semiconductors.³ Group-IV elemental layers are found to exhibit zero band-gap which limits their applications in the semiconductor industry, while their functionalization leads to semiconducting behaviour.¹¹

Recently, 2D atomic layer of phosphorus atoms known as black- or α -phosphorene, with inherent band gap ~ 2 eV has emerged as a potential candidate for semiconductor electronics.¹²⁻¹⁴ Phosphorene-based transistor demonstrates high carrier mobility with higher on/off ratio ($\approx 10^4$) at room

temperature.¹⁵ Electronic band gap of black phosphorene is found to vary inversely with thickness of the layers.¹⁶ Also phosphorene exhibits four thermodynamically stable allotropic forms namely black (α -), blue (β -), gamma (γ -) and delta (δ -)phosphorene, out of these, black (α -) and blue (β -)phosphorene are found to be most stable.^{17,18} The various structural phases of phosphorene show tunability in electronic structure on the application of external electric field, strain and pressure.¹⁹⁻²¹

It is well known that electronic properties change significantly when one goes from 2D monolayer to 1D nanoribbons (NRs). Note that in laboratory situation graphene nanoribbons have been fabricated in controlled fashion by various synthesis techniques.²²⁻²⁵ The electronic properties of phosphorene NRs show strong dependence on their width and edge structure.²⁶⁻²⁸ The electronic band gap of NRs changes on functionalization of edges with various functional groups.^{29,30} Also it is found that electronic band gap varies inversely to the width of NRs.²⁶ Furthermore, electronic properties are strongly influenced by the orientation of nanoribbons.³¹ Mechanical strain and perpendicular electric field also modify the electronic band gap of black phosphorene NRs.^{31,32}

Interestingly, ultra-narrow width armchair graphene NRs as narrow as five carbon atoms, have been successfully synthesized and characterized by the low temperature scanning tunnelling microscopy.³³ Ultra-narrow ribbons have been found to show nearly metallic nature with delocalized orbitals. These ultra-narrow width ribbons have been suggested to be used as

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interconnects in devices in place of Cu. Similarly, ultra-narrow graphene bilayer nanoribbons obtained by unzipping of multiwall carbon nanotube have been characterized under high magnetic field which show Landau quantization.³⁴

Although numerous studies have been made to understand the various properties of black phosphorene nanoribbon,^{26–31} with less attention paid to blue phosphorene based nanoribbons. Here we present a detailed study of ultra-narrow blue phosphorene nanoribbons of varying width with armchair and zigzag edge configurations. The effect of width and mechanical strain on the electronic and dielectric properties of ultra-narrow blue phosphorene NRs have been discussed. The effect of hydrogen passivation at the edges of the considered NRs is also investigated in detail.

2. Computational methods

First principles calculations are performed within density functional theory framework as implemented in SIESTA (Spanish Initiative for Electronic Simulations with Thousands of Atoms) simulation package.^{35–37} Norm conserving Troullier Martin pseudo-potential in fully separable Kleinman and Bylander form has been used to treat the electron–ion interactions.³⁸ The exchange and correlation energies have been treated within the General Gradient Approximation (GGA) according to the Perdew–Burke–Ernzerhof (PBE) parameterization.³⁹ The Kohn Sham orbital's were expanded in a linear combination of numerical pseudo atomic orbital's using a split-valence double zeta basis set with polarization functions (DZP).

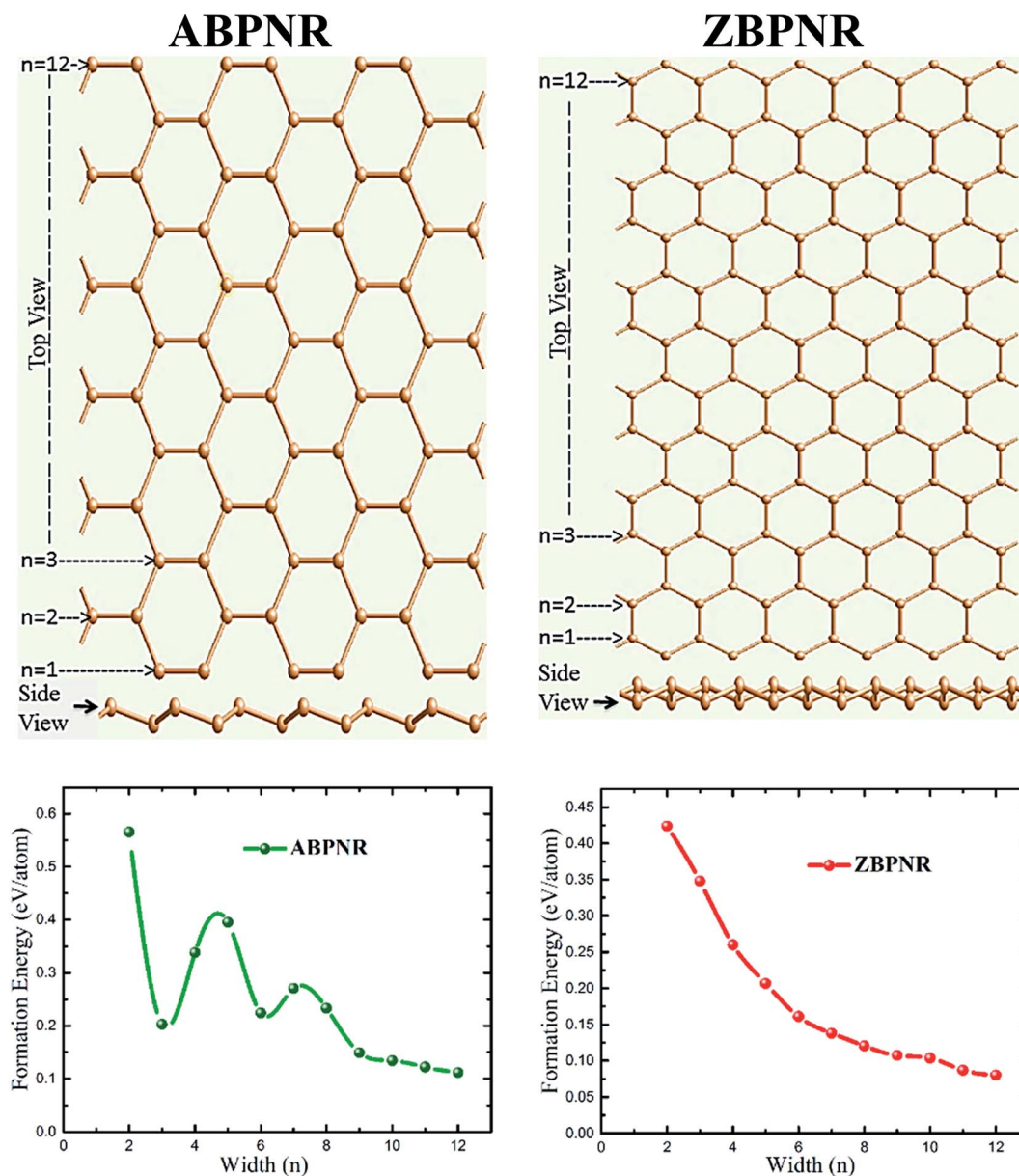


Fig. 1 Top and side view of armchair blue phosphorene nanoribbon (ABPNR) and zigzag blue phosphorene nanoribbon (ZBPNR), where ' n ' represent the number of atomic lines in nanoribbons (top panel). Formation energy as a function of width is plotted in bottom panel.



Throughout geometry optimization confinement energy of numerical pseudo-atomic orbital's are taken as 0.01 Ry. Minimization of energy was carried out using standard conjugate-gradient (CG) technique. Structures were relaxed until the forces on each atom were less than 0.01 eV \AA^{-1} . Monkhorst–Pack scheme is used to sample Brillouin zone with a $10 \times 1 \times 1$ mesh. The spacing of the real space used to calculate the Hartree, exchange and correlation contribution of the total energy and Hamiltonian was 450 Ry. The distance between the periodic images in non-periodic direction of ribbons is kept more than 10 \AA to avoid the mutual interaction between the images.

Optical broadening of 0.2 eV was used for dielectric properties calculations. All the unoccupied states have been considered for the calculations of optical spectra. The imaginary part of dielectric function (ϵ_2) has been estimated by calculating the dipolar transition matrix elements between occupied and unoccupied single electron eigenstates.³⁵ The real part of dielectric function (ϵ_1) has been obtained using Kramers–Kronig transformations. Finally, the Electron Energy Loss

Spectra (EELS) was calculated using ϵ_1 and ϵ_2 in the following formula:

$$\text{Im}\left(-\frac{1}{\epsilon(\omega)}\right) = \frac{\epsilon_2(\omega)}{\epsilon_1^2(\omega) + \epsilon_2^2(\omega)} \quad (1)$$

3. Results and discussions

Blue phosphorene possess graphene-like honeycomb structure with 0.59 \AA out-of-plan buckling of P-atoms. Our GGA-PBE calculated lattice constant 3.31 \AA is in excellent agreement with previously reported value 3.33 \AA .^{17,18} The cohesive energy of blue and black phosphorene was found to be nearly same ($\Delta E \sim 1 \text{ meV}$ per atom),¹⁷ suggesting both allotropic layers to be equally stable. Our calculated indirect band gap of 2.02 eV is also in excellent agreement with the previously reported value $\sim 2 \text{ eV}$ at GGA-PBE level of theory.^{17,18}

Similar to graphene NRs, blue phosphorene nanoribbons (BPNRs) can be obtained by cutting monolayer blue phosphorene along armchair (AC) or zigzag (ZZ) directions. The

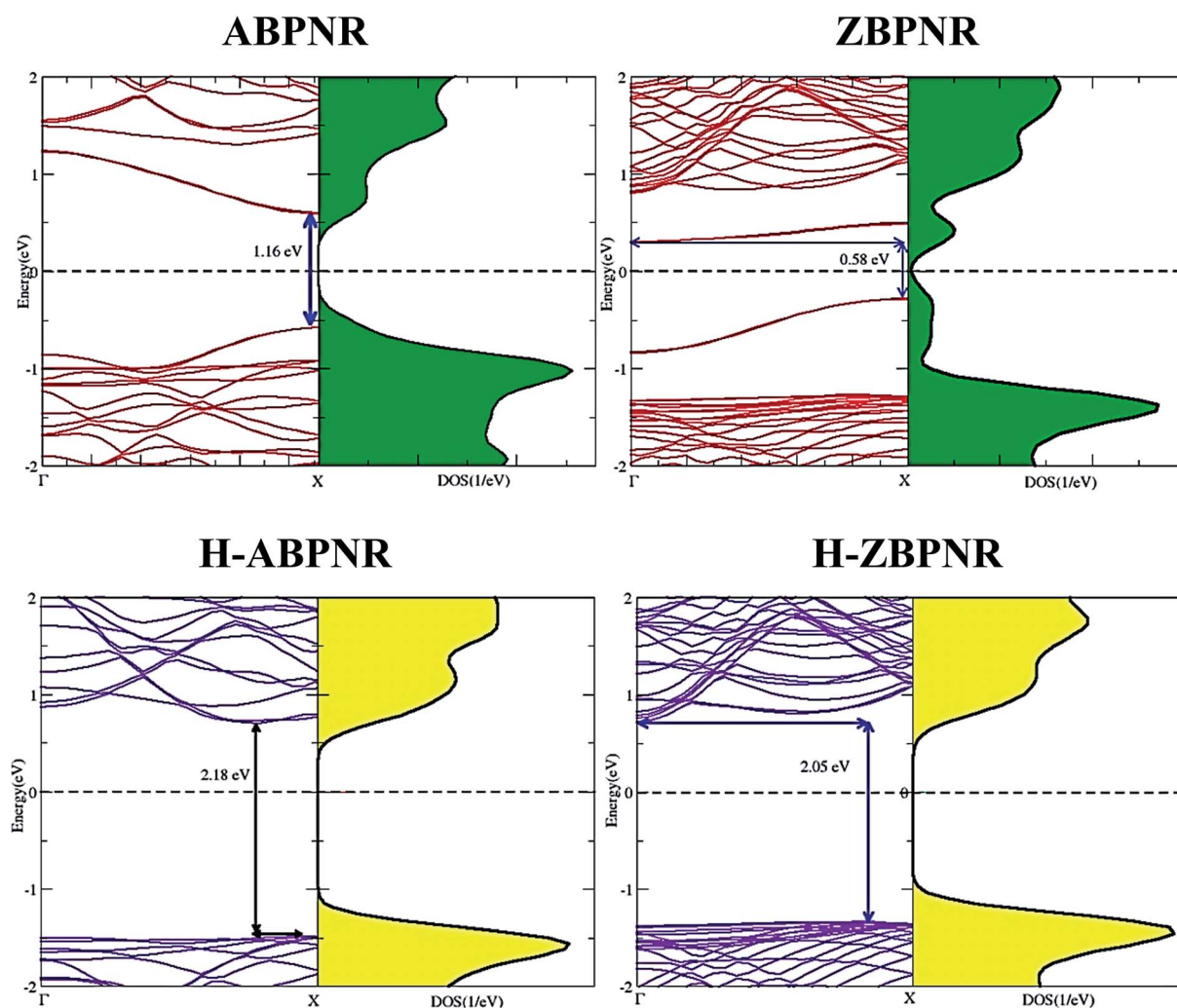


Fig. 2 Electronic band structure and corresponding density of states of armchair blue phosphorene nanoribbon (ABPNR), zigzag blue phosphorene nanoribbon (ZBPNR) and hydrogen passivated edges structures (H-ABPNR and H-ZBPNR) at width $n = 12$.



width of the armchair blue phosphorene nanoribbon (ABPNR) and zigzag blue phosphorene nanoribbon (ZBPNR) can be identified by counting number of atomic lines (n) of P atoms across the ribbon. The ball and stick model of ABPNR and ZBPNR is shown in Fig. 1, where we have considered width (n) from 2 to 12. Note that various configurations of edge structures

are possible in nanoribbons,^{40–44} however, armchair and zigzag edges are most common stable configurations for graphene-like nanoribbons. In case of graphene nanoribbons, zigzag-57 (stone-wales defected) edge structure were reported to be most stable with very small energy difference ($0.02 \text{ eV } \text{Å}^{-1}$) relative to armchair edge structure.⁴² To introduce stone wales defects, one

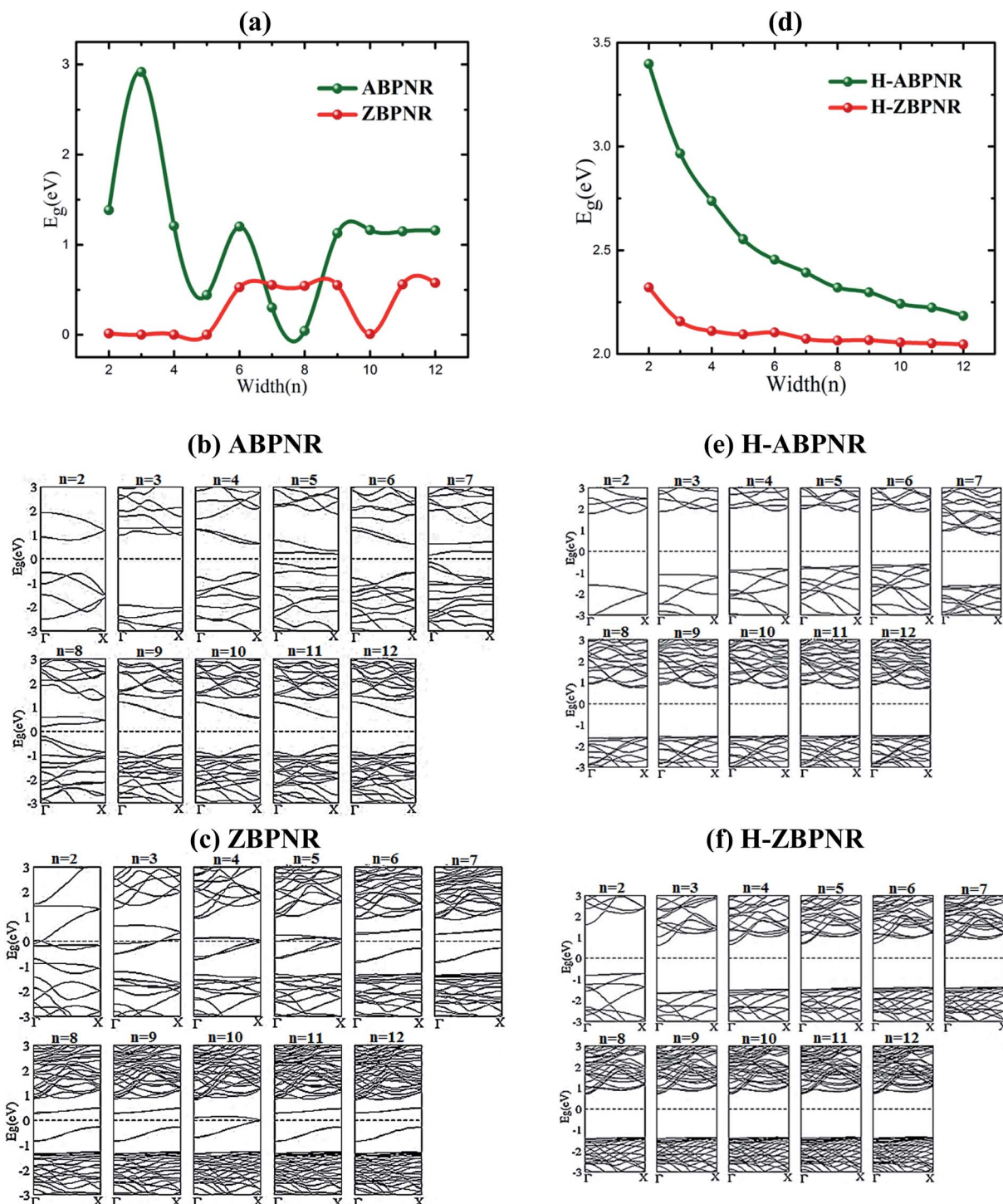


Fig. 3 Electronic band gap (E_g) as a function of width (n) and corresponding band structures for bare (a–c) and edge passivated (d–f) nanoribbons.



need to consider wider and long ribbon to mimic the structural defects in nanoribbons.

Furthermore, the edges are passivated with hydrogen atoms as shown in Fig. S1 of ESI†. On structure relaxation of bare ribbons, the atoms on the edges of both ABPNR and ZBPNR get reconstructed. The reconstruction of atoms for lower width ribbons ($n < 6$) is found to be more pronounced than wider ribbons (Fig. S2 and S3 of ESI†). In order to investigate the relative stability of BPNRs, we have computed the formation energy of both ABPNRs and ZBPNRs. The formation energy (E_F) is defined as:

$$E_F = \frac{E_{\text{BPNR}} - nE_P}{n} \quad (2)$$

where E_{BPNR} is the calculated total energy of given BPNR, E_P is the energy per atom of 2D phosphorene and n is the number of P atoms in BPNR. Formation energy of ABPNR and ZBPNR as a function of ribbon width (n) has been plotted in Fig. 1. The small but positive formation energy indicates that the one-dimensional (1D) nanoribbons are energetically less favourable than the 2D phosphorene counterpart. Formation energy *versus* width shows oscillatory behaviour for ABPNR at lower values which goes away with increase in ribbon width beyond $n = 8$. On the other hand, the formation energies of ZBPNR shows

a systematic decrease in magnitude on increasing ribbon width [Fig. 1]. The formation energy of both ABPNR and ZBPNR at $n = 12$ is calculated as ~ 0.1 eV per atom, indicating that the wider ribbons have relatively higher stability.

Similar to monolayer blue phosphorene, both ABPNR and ZBPNR exhibit semiconducting character, however, with smaller band gap. The electronic band structure of ABPNR at $n = 12$ show 1.16 eV direct band gap while ZBPNR possess indirect band gap 0.58 eV at the same width at GGA-PBE level of theory [Fig. 2]. The states around the Fermi energy are due to the dangling bonds present at the edges of nanoribbons. The band gap for H-passivated edges of ABPNR and ZBPNR is calculated as 2.18 eV and 2.05 eV, respectively, using GGA-PBE functional [Fig. 2].

3.1 Width-dependent electronic properties of BPNRs

Electronic properties of both ABPNR and ZBPNR show distinct behaviour on varying the width (n) of ribbons, *e.g.*, the electronic band gap of ABPNR as function of width show oscillatory behaviour for $n \leq 9$, while at higher width the band gap remain nearly constant (~ 1 eV) (Fig. 3(a)); ZBPNR show metallic characteristics for $n \leq 5$ and semiconducting features ($E_g \sim 0.1$ eV) at higher width ($n = 6-9$). The oscillatory behaviour in the

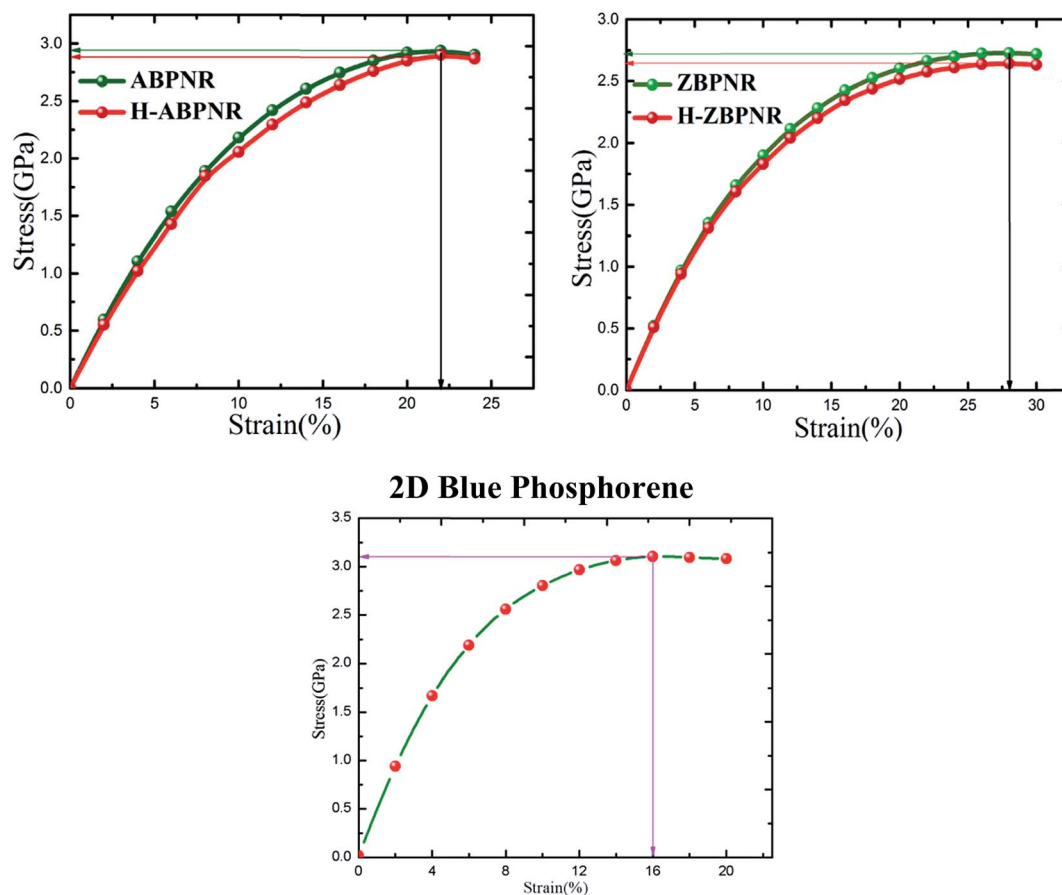


Fig. 4 Strain *versus* stress curves of bare and hydrogen passivated edge structure of blue phosphorene nanoribbons for width $n = 12$. The horizontal and vertical lines with arrow indicate the tensile strength and ultimate tensile strain of nanoribbons. Strain–stress curve for 2D blue phosphorene has also been given for comparison.



electronic band gap for both kind of ribbons may be attributed to the reconstruction of edge atoms as shown in Fig. S2 & S3 of ESI,† which is more pronounced for lower width bare BPNRs. It is evident from Fig. 3(b) and (c) that dispersion of bands around Fermi energy due to unsaturated bonds gets modified significantly on varying the width that results into change in the electronic behaviour of ribbons. Note that the oscillatory behaviour of electronic band gap of nanoribbons at lower width has also been observed in previous studies.^{26–28}

On passivating the dangling bond states with hydrogen atoms at the edges of both kind of ribbons, the band gap shows increase in its magnitude (Fig. 3(d)). The band gap of H-ABPNR shows systematic decrease from 3.4 eV at $n = 2$ to 2.3 eV at $n = 12$. Similarly, the band gap of H-ZBPNR decreases with width, however, the change in the band gap value is found to be very small (~ 0.3 eV) on increasing width from $n = 2$ to $n = 12$. The band gap value of both H-ABPNR and H-ZBPNR nearly converges at higher width of ribbons. It is also evident from Fig. 3(e) and (f) that the dangling bond states disappears on passivation of edge atoms which results into increase in the band gap of both kind of ribbons as compared to bare BPNRs, while it decreases with increasing the width of ribbons.

In order to get further insight into the change in electronic band gap of BPNRs, we calculate the valance band maximum (VBM) and conduction band minimum (CBM) charge density (Fig. S4 and S5 of ESI†). For smaller width the VBM and CBM charge densities are localized on each atoms. As we increase the width, both VBM and CBM charge density show small polarization which is more pronounced at higher width that lifts the VBM and CBM towards Fermi level. The redistribution of charge with changing the width of BPNRs leads to the change in electronic band gap.

3.2 Effect of mechanical strain in BPNRs

In order to estimate the breaking strength of BPNRs, we now apply longitudinal uniaxial strain to the nanoribbons. Since the wider ribbons are found to be energetically more stable, therefore, the strain was applied to the widest ribbon *i.e.* $n = 12$ considered in this study. Mechanical strain ($\frac{\Delta a}{a_0}$) was applied in small steps where a_0 is equilibrium lattice constant and Δa is the change in lattice constant obtained after uniaxial deformation of 1D lattice. The maximum stress that an 1D system can withstand before breaking gives its ultimate tensile

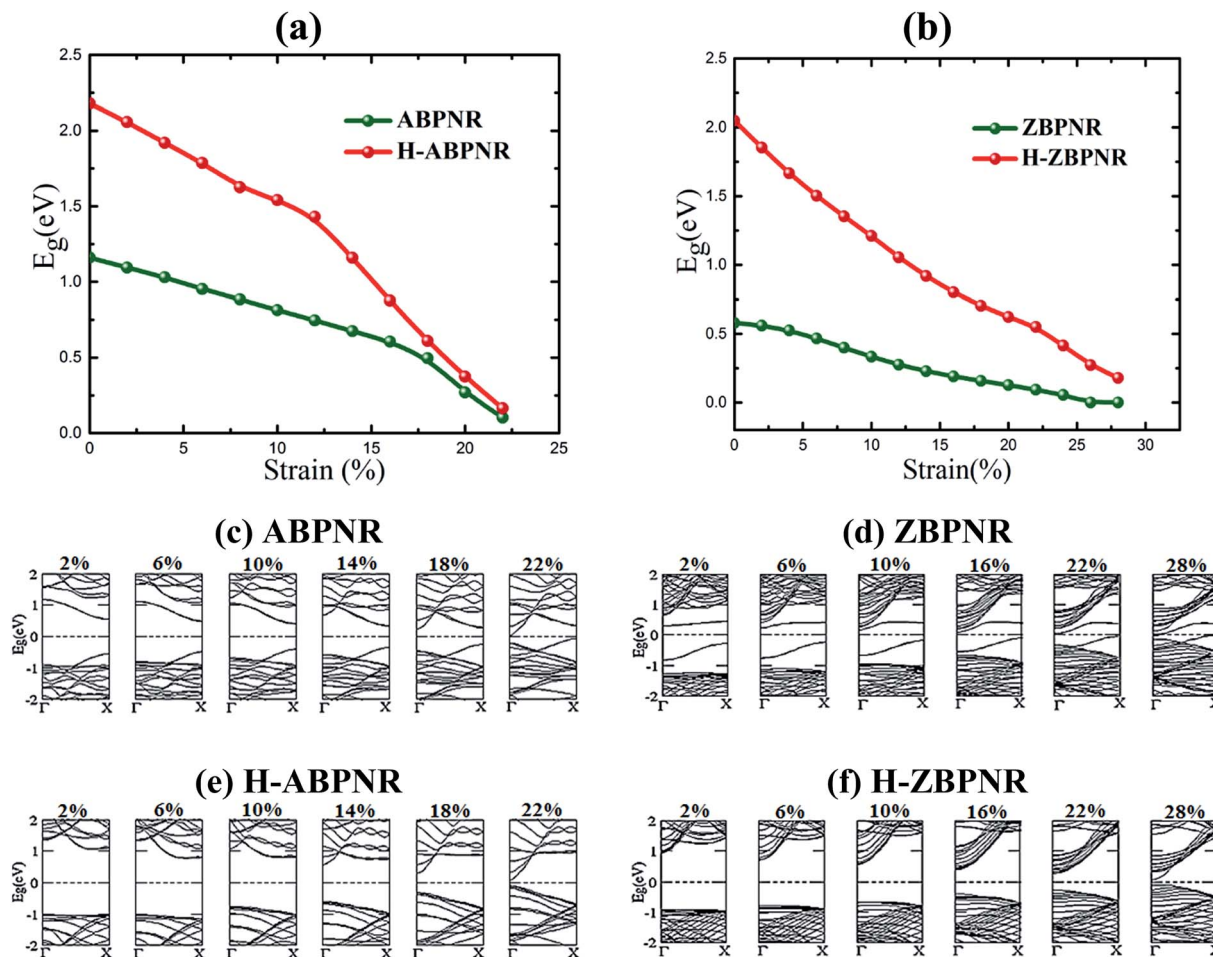


Fig. 5 (a and b) Variation in band gap with applied strain for bare and hydrogen passivated edges of blue phosphorene nanoribbons for width $n = 12$. (c–f) Electronic band structures at different strain values are also shown.



strength (UTS). The value of UTS is the point at which the slope of the strain–stress curve becomes zero and the critical value of strain at that point is the ultimate strain applicable to 1D system. The strain–stress has been determined by calculating the stress tensor components in response to the strain tensor. The stress tensor is defined as a positive derivative of total energy with respect to the strain tensor.³⁵

The ultimate strain that can be applied to ABPNR and ZBPNR without breaking is calculated as 22% and 28%, respectively, which is found to be same for H-ABPNR and H-ZBPNR (Fig. 4). UTS for ABPNR and ZBPNR is calculated as ~ 3 GPa and ~ 2.7 GPa, respectively. Note that ultimate strain and UTS for 2D blue phosphorene has been calculate as 16% and 3.1 GPa, respectively (Fig. 4). The UTS remains nearly same as we go from monolayer blue phosphorene to ultra-narrow BPNR while ultimate strain increases from 16% to more than 20%. It is noticeable that despite strong dependence of edge passivation in electronic band structure, the UTS for both bare and passivated BPNRs remains nearly the same. It is worth mentioning here that the uniaxial strain induces in-plan stress

tensor components perpendicular to the direction of applied strain which is found to be negligible as compared to the stress components along strain direction, hence, not given in Fig. 4.

Thereafter, keeping in view the critical value of strain, we investigated the evolution of electronic structure of BPNRs on the function of strain. It is found that the electronic band gap of considered ribbons decreases continuously with increasing strain, eventually, rendering them nearly metallic at a critical value (Fig. 5). On carefully looking at the evolution of electronic structure with strain, the valance band maximum (VBM) and conduction band minimum (CBM) systematically shifts towards Fermi level which tends to close the band gap at Γ -point. At ultimate tensile strain, the semiconductor-to-metal transition occurs for bare ZBPNR while few meV band gap remains intact for other ribbons (Fig. 5). These changes in electronic structure are attributed to the redistribution of atomic orbitals on the application of applied strain.

In order to get deep insight into the change in electronic structure with applied strain, we calculate the orbital resolved partial density of states (PDOS) of unstrained and strained

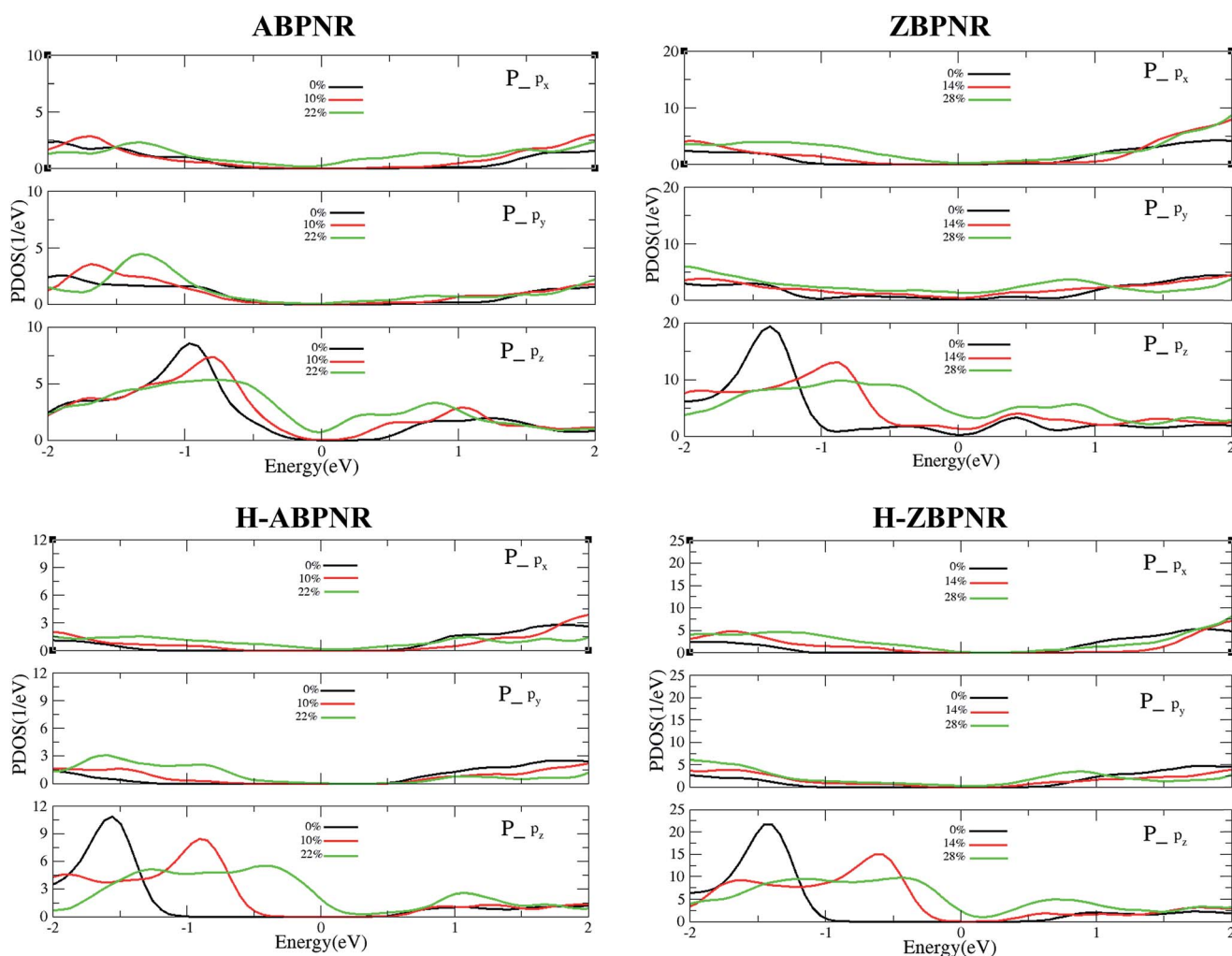


Fig. 6 Orbital resolved partial density of states of in-plan (p_x and p_y) and out-of-plan (p_z) orbitals of phosphorous atoms of strain-free and strained BPNRs.



BPNRs (Fig. 6). It is important to note that on the application of in-plane strain, the out-of-plane orbitals get redistributed in significant manner while in-plane orbitals remain nearly unaffected. The p_z orbitals of phosphorous atom of strained BPNRs shift towards Fermi energy that lead to change in electronic band gap. The shift of p_z orbitals towards Fermi energy is more pronounced for higher strain which results into semiconductor-to-metal transitions in BPNRs at critical value of strain (Fig. 6).

3.3 Dielectric properties of BPNRs

We now investigate the effect of width and mechanical strain on the dielectric properties of BPNRs. The static dielectric constant (ϵ_s) was obtained by calculating the value of real part of dielectric function (ϵ_1) at zero frequency. The calculated ϵ_s as a function of width of both ABPNR and ZBPNR show oscillatory behaviour (Fig. 7a). On careful examination we found that the change in ϵ_s with varying width shows its link with the change in electronic band gap (E_g) on the same width *e.g.* for ABPNR at $n = 8$, $E_g \sim 0$ (minimum) while corresponding ϵ_s is found to be ~ 4.5 (maximum); the maximum E_g in ABPNR is calculated as ~ 3 eV at $n = 3$ while the ϵ_s is calculated minimum (~ 1.9) at the same

width; for ZBPNR, the E_g remains constant at $n = 6-9$ ($E_g \sim 0.5$ eV) and the corresponding ϵ_s is calculated as $\sim 3.5-3.7$ at the same widths. This type of behaviour of E_g and ϵ_s may be attributed to the Penn's model expression,⁴⁵ where electronic band gap of the semiconductor is roughly related with the static dielectric constant (ϵ_s) as $\epsilon_s \approx 1 + \left(\frac{\hbar\omega_p}{E_g}\right)^2$, where ω_p is plasmonic frequency.

In case of H-ABPNR and H-ZBPNR, E_g systematically decreases as a function of width (Fig. 3d) while corresponding ϵ_s at same widths shows inverse trend with fluctuations at higher widths (Fig. 7b) which may be attributed to the pronounced effect of distinctly different plasmonic excitations at wider width ribbons. ϵ_s is also found to change with applied strain (Fig. 7c and d) which shows nearly inverse trend as compared to E_g with strain (Fig. 5a and b). At the critical value of strain, ϵ_s for ABPNR, H-ABPNR, ZBPNR and H-ZBPNR is calculated as 16.2, 15.1, 9.8 and 5.5 respectively. Higher value of ϵ_s indicate metallic or nearly metallic character of these ribbons.

Now we investigate the change in imaginary part of dielectric function (ϵ_2) and electron energy loss spectra (EELS) as a function of width. It is noticeable that ϵ_2 in passivated BPNRs

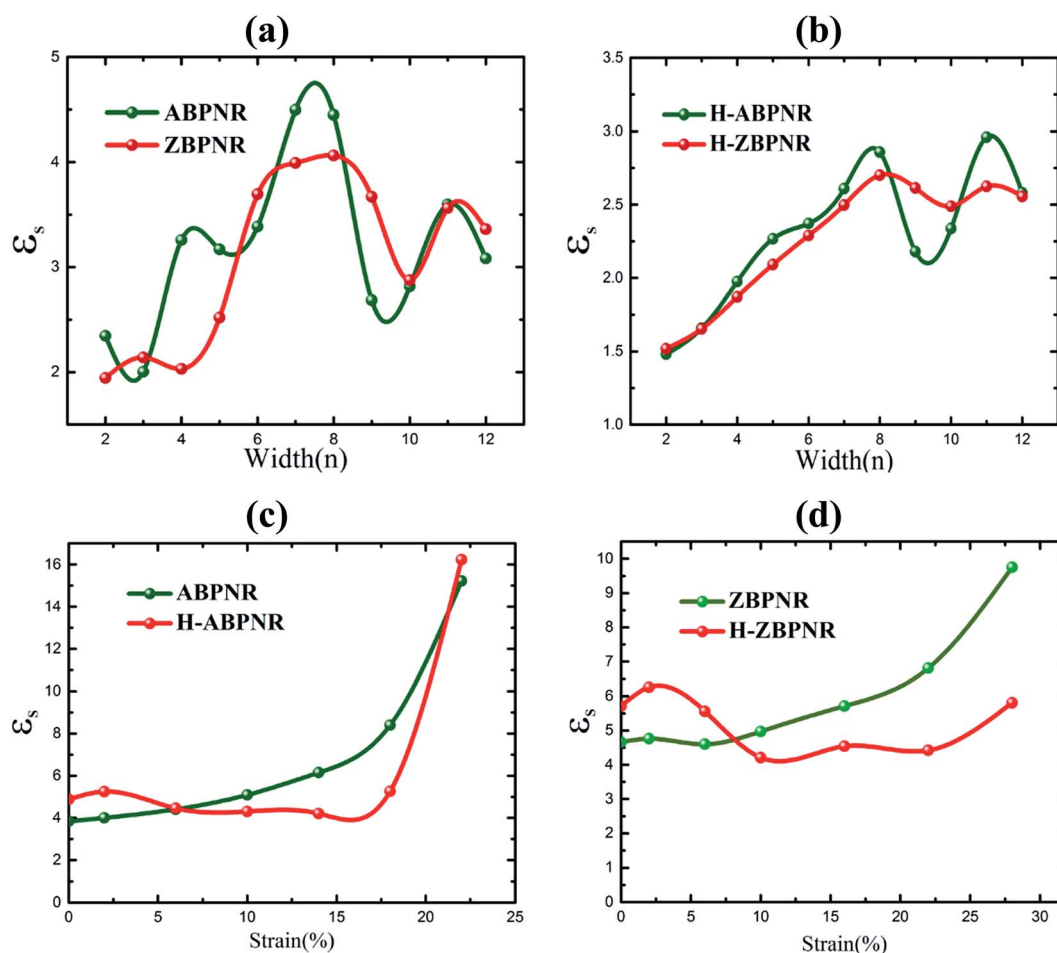


Fig. 7 Static dielectric constant (ϵ_s) versus width (n) for (a) bare and (b) edge passivated BPNRs. ϵ_s as a function of strain for (c) ABPNRs and (d) ZBPNRs are give at width $n = 12$.



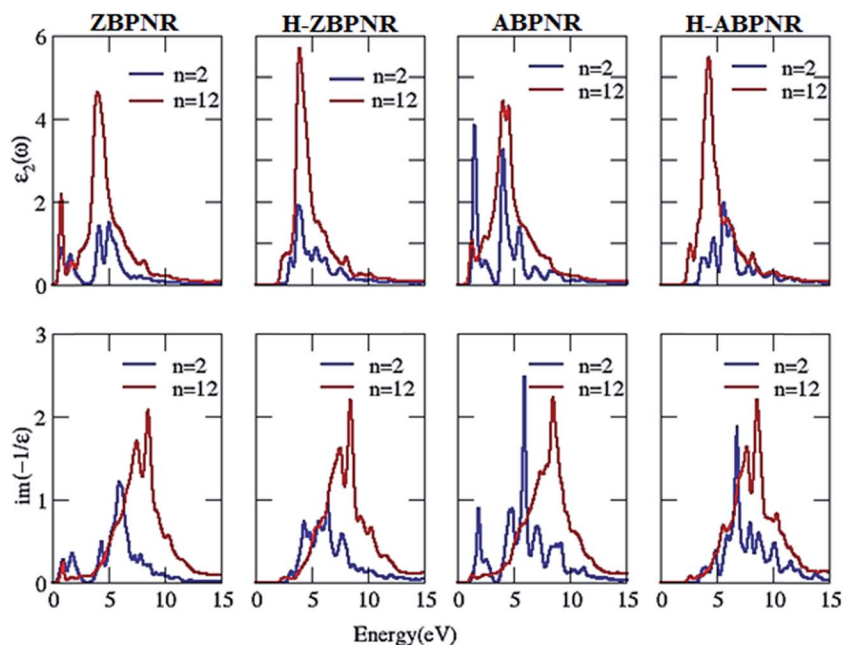


Fig. 8 Imaginary part of dielectric function (ϵ_2) and electron energy loss spectra (EELS) at $n = 2$ and $n = 12$ for bare and hydrogen passivated edges of blue phosphorene nanoribbons.

possess structure peak at ~ 5 eV while additional structure at ~ 1 eV is found for bare BPNRs (Fig. 8). We attribute the additional structure in bare BPNRs to the interband transitions between the dangling bond states near Fermi energy which disappear in H-BPNRs. The energy of structure peaks of ϵ_2 remains independent of width while the intensity of wider width BPNRs is at higher magnitude. The higher intensity of

wider BPNRs is due to the increased number of allowed bands available for interband transitions.

The structure peaks in electron energy loss spectra (EELS) spectra give the collective excitation of electron which is known as plasmons. Low energy plasmons are called π -plasmons which are associated with the excitation of weak π -electrons while high energy plasmons are $\pi + \sigma$ plasmons which are

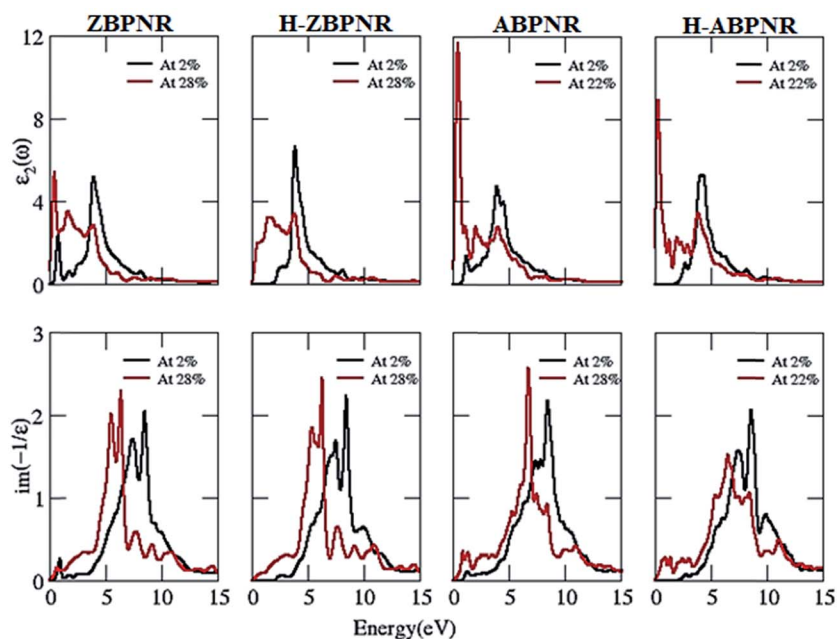


Fig. 9 Imaginary part of dielectric function (ϵ_2) and electron energy loss spectra (EELS) at 2% and ultimate tensile strain for bare and hydrogen passivated edges of blue phosphorene nanoribbons.



associated with both weak π and strong σ electrons.^{46,47} EELS show two plasmonic features for bare BPNRs, one at low energy <5 eV and second at higher energy >5 eV, while H-BPNRs have been found to exhibit plasmonic excitations only at higher energy (>5 eV) (Fig. 8). The low energy plasmonic features (<5 eV) are due to π -electrons while the higher energy (>5 eV) plasmonic features comes from both π - and σ -electrons of BPNRs. Interestingly, plasmonic energy gets blue shifted as we go from width $n = 2$ to $n = 12$ (Fig. 8). The blue shift in the plasmonic energy is calculated as large as ~ 3 eV (Table T1 of ESI†). This much amount of plasmonic energy difference makes electron energy loss spectroscopy useful in order to identify the width of BPNRs in real experimental situation.

Next we consider the effect of mechanical strain on ϵ_2 and EELS of BPNRs. The uniaxial strain induces small red shift in ϵ_2 of all the considered BPNRs (Fig. 9), which is attributed to the modification in electronic band dispersion due to superposition of atomic orbitals on the application of applied strain. Similarly, EELS of BPNRs is also found to be red shifted on the application of strain. The red shift in EELS is more pronounced in $\pi + \sigma$ plasmonic structures (Fig. 9) where as large as ~ 2 eV shift is calculated in plasmonic energy (Table T1 of ESI†). These tunable dielectric properties of BPNRs with width and external strain may have importance in optoelectronic devices based on BPNRs.

4. Summary

In summary, we have performed a detailed first principles study of optoelectronic properties of ultra-narrow BPNRs. The electronic band gap of BPNRs is found to be width and edge-specific. The ultimate tensile strength of BPNRs is calculated as ~ 3 GPa with critical strain value >20%. BPNRs are found to possess nearly metallic characteristics at critical value of strain which is consistent with the higher value of calculated static dielectric constant. ϵ_2 for passivated BPNRs possess structure peak at ~ 5 eV while additional structure at ~ 1 eV is found for bare BPNRs which is attributed to the interband transitions between the dangling bond states near the Fermi energy. Interestingly, plasmonic energy gets blue shifted on increasing the width of BPNRs while it shows red-shift on the application of longitudinal tensile strain. These tunable electronic and dielectric properties of BPNRs may be useful for the optoelectronic applications.

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