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A facile sacrificial template method to synthesize one-dimensional porous CdO/CdFe₂O₄ hybrid nanoneedles with superior adsorption performance†

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In this work, we have developed a facile sacrificial template method to prepare one-dimensional (1D) porous CdO/CdFe₂O₄ hybrid nanoneedles (HNNs) with uniform pore size distribution and structural stability for the first time. The solid 1D Cd₂Fe(CN)₆ NNs with a smooth surface are first synthesized and used as sacrificial templates through a simple low-temperature hydrothermal process. After thermal treatment, the solid precursor NNs can be transformed into the corresponding porous CdO/CdFe₂O₄ HNNs. Due to the large-scale production and unique structural features, the as-prepared porous CdO/CdFe₂O₄ HNNs can serve as a highly efficient adsorbent towards removal of Congo red (CR) from aqueous solution with a maximum adsorption capacity of 1491 mg g⁻¹. Furthermore, the adsorption process of CR on porous CdO/CdFe₂O₄ HNNs was systematically carried out, indicating that the Langmuir model and the pseudo-second-order kinetics model can describe well the adsorption behavior. The result demonstrates that the as-prepared porous CdO/CdFe₂O₄ HNNs may be a promising candidate in dye effluent treatment.

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1. Introduction

Over the last few years, water pollution has received much attention and become a global problem, because organic dyes discharged from the textile industry are the main source of sewage.^{1,2} As a class of synthetic colored organic compounds and one of the most important benzidine-based anionic dyes, Congo red (CR) is widely adopted in the textile industry. Due to being harmful to many organisms including humans, it is of importance to rationally and efficiently deal with its contaminated effluents.³ Up to now, a great number of methods have been adopted, such as oxidation,⁴ electrochemical degradation,⁵ photocatalysis,^{6–8} coagulation,⁹ adsorption,¹⁰ nanofiltration,¹¹ *etc.* Among these methods, adsorption has been considered as the main method dealing with organic dyes in wastewater, because of its operational ease, low capital and operational costs and high efficiency.^{12–14} Conventional adsorbents, including active carbon,^{15,16} synthetic resins,¹⁷ *etc.*, are limited due to their low adsorptive efficiency and the difficult regeneration and disposal. Therefore, exploring high efficiency

and low cost adsorbent materials is of vital importance for practical application.

Recently, numerous efforts have been made to develop novel low-dimensional or porous adsorbents owing to their small particle sizes, great specific surface areas, porosity, and abundant active sites.^{1,18–26} For example, one-dimensional (1D) porous ZnO/SnO₂ hetero-nanofibers have been successfully prepared *via* a electrospinning method,²⁷ which exhibited the superior adsorption capacity and the enhanced photoactivity for the degradation of CR. Sun *et al.* reported a novel adsorbent based on the porous Al-doped ZnO nanosheets, demonstrating the improved adsorption capacity for dye.²⁸ The synergistic effect of low-dimensional morphology and porous structure can provide a large surface-to-volume ratio and plenty active sites, which can facilitate the diffusion and transport of organic dyes in aqueous solution, resulting in the enhanced adsorptive performance.

Herein, a novel adsorbent based on 1D porous CdO/CdFe₂O₄ hybrid nanoneedles (HNNs) has been synthesized *via* a facile sacrificial template method for the first time. The solid 1D Cd₂Fe(CN)₆ NNs are first synthesized and used as sacrificial templates by a simple low-temperature hydrothermal route, and then the solid precursor NNs can be transformed into the corresponding porous 1D CdO/CdFe₂O₄ HNNs after thermal treatment. Because of the large-scale production and unique structural features, the as-obtained 1D porous HNNs exhibit an superior adsorption capacity of 1491 mg g⁻¹ for CR from

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aqueous solution. The adsorption process of CR on 1D porous CdO/CdFe₂O₄ HNNs was systematically carried out, which is in accordance with the pseudo-second-order kinetics model and Langmuir isotherm. Further investigations show that the adsorption process is endothermic and spontaneous, and the adsorption mechanism is attributed to a physisorption process.

2. Experimental section

2.1 Synthesis of Cd₂Fe(CN)₆ NNs and porous CdO/CdFe₂O₄ HNNs

All reagents used in this work were of analytical grade and purchased from the Shanghai Chemical Reagent Factory. In a typical synthesis, 22.8 mg of CdCl₂·2.5H₂O and 0.4 g of polyvinylpyrrolidone (PVP, MW ~58k) were dissolved in 20 mL of distilled water to form a clear solution. Then, 20 mL of pre-prepared K₄Fe(CN)₆·3H₂O aqueous solution (2.5 mM) was added dropwise into the above solution forming a white colloidal solution and kept stirring for 10 min. The resulting mixture was then transferred into a 50 mL Teflon-lined stainless steel autoclave, which was sealed and heated in an oven at 120 °C for 20 h and then cooled to room temperature naturally. After collection by centrifugation, the resultant white products (1D Cd₂Fe(CN)₆ NNs) were washed with anhydrous ethanol and distilled water for three times and then dried under vacuum at 80 °C for 4 h. Finally, the 1D porous CdO/CdFe₂O₄ HNNs were obtained by annealed in air at 550 °C maintaining for 4 h with a ramp rate of 1 °C min⁻¹ using the as-prepared Cd₂Fe(CN)₆ NNs as precursor.

2.2 Characterizations

Powder X-ray diffraction (XRD) measurements of the samples were performed with a Philips PW3040/60 X-ray diffractometer using Cu K α radiation at a scanning rate of 0.06 deg s⁻¹. Scanning electron microscopy (SEM) was taken with a Hitachi S-4800 scanning electron micro-analyzer with an accelerating voltage of 15 kV. Transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) were conducted using a JEM-2100F field-emission TEM. The adsorption spectra were measured using a PerkinElmer Lambda 900 UV-vis spectrophotometer at room temperature. N₂ adsorption-desorption isotherms were carried out at 77 K on a Micrometrics ASAP 2020 surface area and porosity analyzer after the sample had been degassed in vacuum at 160 °C for 4 h.

Fourier transform infrared spectra (FT-IR) were recorded on a Nicolet NEXUS670 FT-IR spectrometer using KBr pellets.

2.3 Adsorption performance test

For adsorption isotherm studies, 0.02 g of 1D porous CdO/CdFe₂O₄ HNNs adsorbents were added to 50 mL of CR with different concentrations solution (60, 200, 300, 400, 500, 600, 800, 1000 mg L⁻¹, respectively.) in a conical flask under stirring at a natural pH value and room temperature (298 K) for 8 h. After adsorption, the resulting suspensions were collected and separated to measure the adsorption peak intensity of CR (at 496 nm) by using UV-vis spectroscopy. To study adsorption kinetic, 0.02 g of the CdO/CdFe₂O₄ sample was mixed with 50 mL of CR solution (60, 100, 150 mg L⁻¹) at certain time intervals. Each test was repeated for three times to ensure the accuracy of this work.

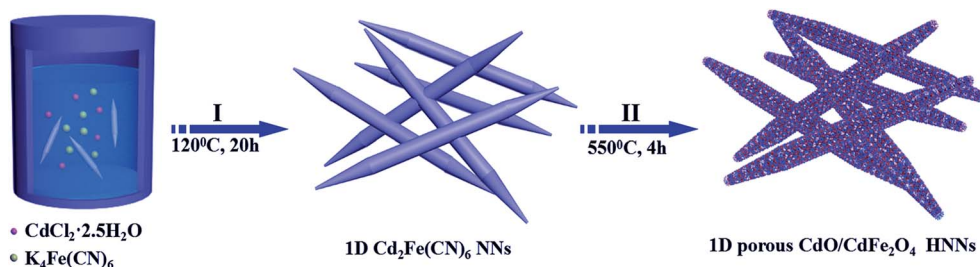
3. Results and discussion

3.1 Synthesis and characterization of 1D porous CdO/CdFe₂O₄ HNNs

The synthesis procedure for the 1D porous CdO/CdFe₂O₄ HNNs is schematically illustrated in Scheme 1, which can be described by two steps. In step I, the uniform 1D Cd₂Fe(CN)₆ NNs as the precursors are prepared *via* a facile low-temperature hydrothermal process at 120 °C for 20 h. In step II, the solid precursors can be transformed into the corresponding 1D porous CdO/CdFe₂O₄ HNNs after thermal treatment at 550 °C for 4 h. The pores formed in the CdO/CdFe₂O₄ HNNs by generating gases from the interior of the precursor during the thermal decomposition process can be described as the following reaction equation (eqn (1)), which is much similar to the previous literature.²⁹



The XRD analysis was first carried out to investigate the crystallographic structure and phase purity of the as-prepared 1D Cd₂Fe(CN)₆ NNs and porous CdO/CdFe₂O₄ HNNs. From the XRD pattern of the as-prepared precursor (Fig. S1, see ESI[†]), all of the observed peaks can be ascribed to the characteristic peaks of cadmium iron cyanide (Cd₂Fe(CN)₆, JCPDS card no. 24-0164). Fig. 1a shows the XRD pattern of the as-prepared product after thermal treatment at 550 °C. The pattern obviously consist of two sets of diffraction peaks, the diffraction peaks marked by asterisk correspond to the cubic phase of CdO ($a = b = c = 4.695$



Scheme 1 Schematic illustration of the formation of 1D porous CdO/CdFe₂O₄ HNNs.



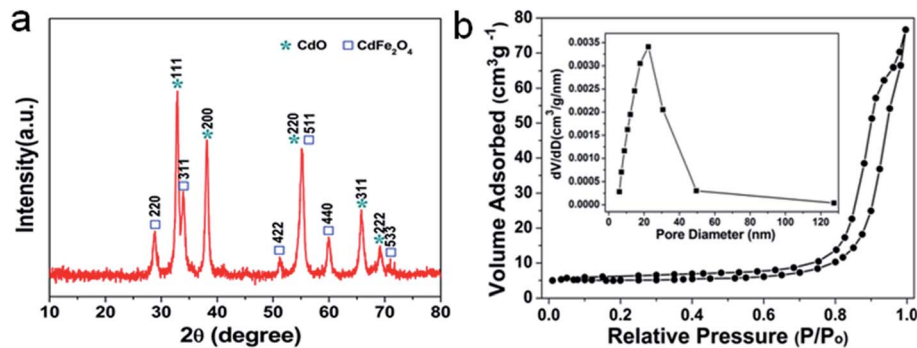


Fig. 1 (a) XRD pattern and (b) N_2 adsorption–desorption isotherm and pore-size distribution curve (inset) of the as-obtained 1D porous CdO/CdFe $_2$ O $_4$ HNNs.

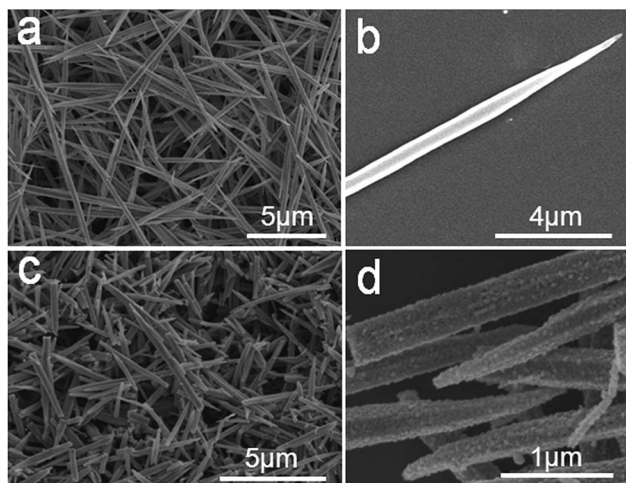


Fig. 2 SEM images of the as-prepared (a) and (b) 1D Cd $_2$ Fe(CN) $_6$ NNs precursor; (c) and (d) 1D porous CdO/CdFe $_2$ O $_4$ HNNs.

Å, JCPDS card no. 05-0640), and the other peaks marked by square match well with the cubic phase of CdFe $_2$ O $_4$ ($a = b = c = 8.700$ Å, JCPDS card no. 22-1063). No impurity peaks are detectable, indicating the successful transformation of Cd $_2$ -Fe(CN) $_6$ to CdO/CdFe $_2$ O $_4$ hybrids. The porous nature of the as-obtained 1D CdO/CdFe $_2$ O $_4$ HNNs was further assessed by nitrogen adsorption isotherm experiment, as shown in Fig. 1b. It can be observed that the adsorption–desorption isotherm of the as-prepared product presents a type IV isotherm with a distinct hysteresis loop at the relative pressure region ($P/P_0 = 0.7$ – 1), manifesting the existence of plentiful mesopores in the product.^{30–32} The surface area of the 1D porous CdO/CdFe $_2$ O $_4$ HNNs can be found to be 16.2 m 2 g $^{-1}$ and the Barrett–Joyne–Halenda (BJH) pore-size distribution curve (inset in Fig. 1b) display the average size of the pores is centered at approximately 25.3 nm and the pore volume is calculated to be about 0.1188 cm 3 g $^{-1}$. These porous channels are favorable to promote the diffusion of dyes into the adsorbents, resulting in the enhanced adsorption capability for CR.

Fig. 2 shows the SEM images of the as-prepared 1D Cd $_2$ -Fe(CN) $_6$ NNs and porous CdO/CdFe $_2$ O $_4$ HNNs. A panoramic view of the as-obtained Cd $_2$ Fe(CN) $_6$ precursor (Fig. 2a) reveals

that the sample is composed of uniform 1D needle-like nanostructures with a diameter of *ca.* 300 nm and *ca.* 10 μm in length. The high-magnification image (Fig. 2b) of a single Cd $_2$ Fe(CN) $_6$ NN further display the surface of the precursor is rather smooth. Fig. 2c and d display the SEM images of the as-prepared 1D porous CdO/CdFe $_2$ O $_4$ HNNs, which are significantly different from those of the precursor. The rough surface and the disordered pores can be obviously observed, but the 1D needle-like morphology is well preserved, indicating that the as-obtained 1D porous structure is stability. This result also confirms the successful conversion of Cd $_2$ Fe(CN) $_6$ NNs into porous CdO/CdFe $_2$ O $_4$ HNNs.

The structural information and characterization of the as-obtained 1D porous CdO/CdFe $_2$ O $_4$ HNNs was verified by TEM image, as shown in Fig. 3a. It further reveals that needle-like structure has rough surface, which is in agreement with the SEM observations. From the enlarged TEM images (Fig. 3b and c), the 1D needle-like morphology is completely composed of numerous nanoparticles and the porous structure on the surface of HNN can be clearly observed. HRTEM image (Fig. 3d) shows the clear lattice spacings of 0.23 and 0.26 nm are corresponding to the (200) plane of cubic phase CdO and (311) plane of cubic phase CdFe $_2$ O $_4$, respectively. The spatial distribution of the compositions in a single CdO/CdFe $_2$ O $_4$ HNN is also studied by elemental mapping. As shown in Fig. 3e–h, the mapping results show uniform distribution of O, Fe and Cd elements throughout the entire 1D HNN.

3.2 The adsorption capacity of CR

Inspired by the porous structure, the as-prepared 1D CdO/CdFe $_2$ O $_4$ HNNs possess more active adsorption sites and efficient transport pathways, which may contribute to excellent adsorption performance. In this work, we have studied the performance of the as-prepared product as a new type of adsorbent, and CR is chosen as a model organic pollutant. The repeated adsorption isotherm and standard deviations of CR at different initial concentrations are shown in Fig. 4a. The adsorption capacity can be determined using the follow expression (eqn (2)):

$$q_t = (C_0 - C_t)V/m \quad (2)$$



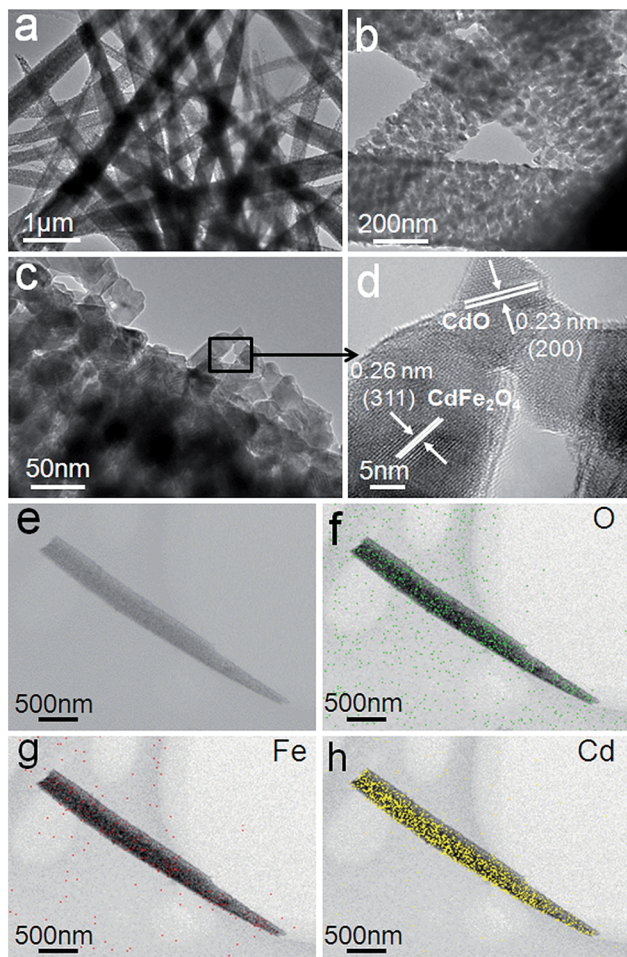


Fig. 3 (a–c) TEM images and (d) HRTEM image of the as-prepared 1D porous CdO/CdFe₂O₄ HNNs. (e) STEM image and (f) O, (g) Fe and (h) Cd elemental mapping images of a single CdO/CdFe₂O₄ HNN.

Here, q_t is the adsorption capacity, C_0 and C_t are the initial and final (after adsorption) concentrations of CR (mg L^{-1}), V is the volume of CR solution (L), and m is the mass of the as-prepared 1D porous CdO/CdFe₂O₄ HNNs (g). It can be found that the removal capacity rises dramatically with the increasing equilibrium concentration of CR ranging from 0 to 56.4 mg L^{-1} , and then the trend turns slowly until to reach the saturation value. It can be explained by the fact that the higher initial CR concentration provides a higher driving force, which could promote dye ions from the solution to the surface of the 1D HNNs, and cause more collisions between active sites and dye ions on the adsorbents. When the initial concentration of CR is 1000 mg L^{-1} , the maximum adsorption capacity can reach ca. 1491 mg g^{-1} . The removal capacities of CR using the porous CdO/CdFe₂O₄ HNNs as adsorbent and other adsorbents reported previously are listed in Table S1 (see ESI†). It can be seen that most of the adsorption capacity of the adsorbent was less 500 mg g^{-1} , and only a few higher than 1000 mg g^{-1} .^{33–38} Therefore, we can conclude that the as-prepared 1D porous HNNs possess superior adsorption performance.

3.3 Adsorption isotherms

Adsorption isotherm provides specific information both the capacity of the adsorbent and the interactions between adsorbent and CR molecules. Establishing the most appropriate correlations for the equilibrium adsorption capacity of the adsorption system is requested. Among of the most commonly applied isotherm theories, the Langmuir (eqn (3)) and Freundlich (eqn (5)) isotherms were usually used to describe this adsorption process. The Langmuir isotherm is based on the hypothesis that the structure of adsorbent is homogeneous, where all adsorption sites are identical and energetically equivalent (eqn (3)).³⁹

$$q_e = \frac{q_m K_L C_e}{1 + K_L C_e} \quad (3)$$

And the linear form can be expressed as follows (eqn (4)):

$$\frac{C_e}{q_e} = \frac{1}{K_L q_m} + \frac{C_e}{q_m} \quad (4)$$

In which C_e and q_e are as mentioned above, q_m (mg g^{-1}) is the theoretical maximum monolayer adsorption capacity and K_L (L mg^{-1}) is the Langmuir adsorption constants related to adsorption rate.

The Freundlich model represents a heterogeneous system and reversible adsorption, which is characterized by a heterogeneity factor of $1/n$. This model is not restricted to monolayer formation (eqn (5)).³⁹

$$q_e = K_F C_e^{1/n} \quad (5)$$

Meanwhile, eqn (5) can also be rearranged to obtain a linear form (eqn (6)):

$$\log q_e = \log K_F + \frac{1}{n} \log C_e \quad (6)$$

For the Freundlich mode, the constant K_F (L mg^{-1}) and n are the Freundlich adsorption constant, K_F is roughly adsorption capacity at a unit concentration, and n is the adsorption intensity.

Both of the Langmuir and the Freundlich models were applied to analyze the equilibrium adsorption data of CR onto porous CdO/CdFe₂O₄ HNNs adsorbent. The plots are shown in Fig. 4b and S2a (see ESI†). Additionally, Table S2 (see ESI†) summarizes the model parameters calculated from the slope and intercept of the both models, as well as the coefficients of determination R^2 . For Langmuir model, the correlation coefficient is $R^2 = 0.99978$, while $R^2 = 0.75854$ for the Freundlich model. Therefore, the adsorption process of CR on the porous CdO/CdFe₂O₄ HNNs obeys Langmuir model, indicating the adsorbed layer is monolayer coverage. The calculated adsorption capacity ($q_{e,\text{cal}}$) from Langmuir model of the product for CR can be estimated to be 1494 mg g^{-1} , which is close to the experimental value (1491 mg g^{-1}).



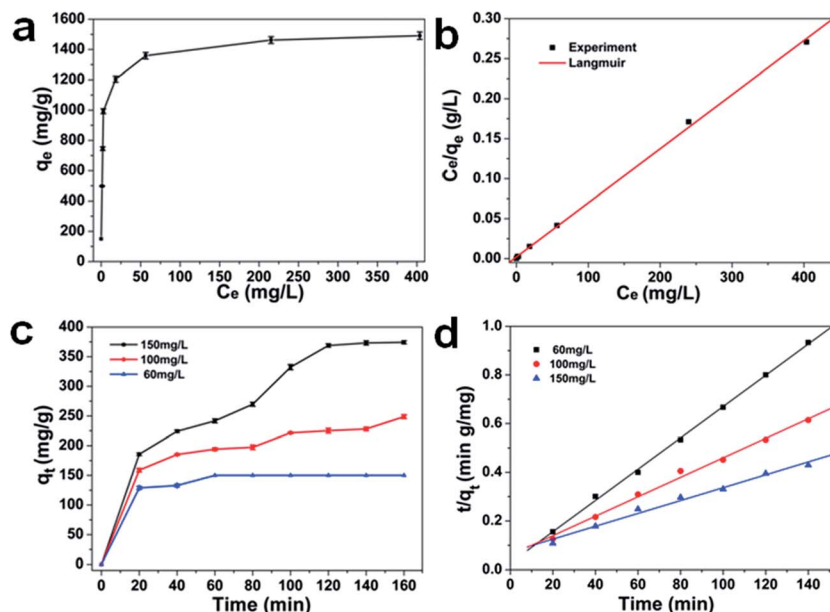


Fig. 4 (a) Equilibrium adsorption isotherms of CR concentrations on porous CdO/CdFe₂O₄ HNNs. (b) The linear dependence of C_e/q_e versus C_e based on the Langmuir isotherm model for different equilibrium concentrations. (c) Time profiles of CR adsorption capacity on the porous CdO/CdFe₂O₄ HNNs (d) the plots of pseudo-second-order kinetics rates for adsorption of CR on the as-obtained porous CdO/CdFe₂O₄ HNNs ($T = 298$ K). The error bar represents the standard deviations ($n = 3$).

3.4 Adsorption kinetics

To study the effect of contact time and evaluate properties, adsorption kinetic experiments were further performed. Fig. 4c shows the effect of contact time at different initial concentrations (60, 100, 150 mg L⁻¹) and standard deviations with 1D porous CdO/CdFe₂O₄ HNNs. From this plot, the adsorption capacity for CR rises with the increase of adsorption time, reaching adsorption equilibrium at about 60, 100 and 120 min for 60 mg L⁻¹, 100 mg L⁻¹ and 150 mg L⁻¹, respectively. Two famous pseudo-first-order (eqn (7)) and pseudo-second-order (eqn (8)) kinetic models³⁹ were used to investigate the characteristics of the adsorption process.

$$\ln(q_e - q_t) = \ln q_e - k_1 t \quad (7)$$

$$\frac{t}{q_t} = \frac{1}{k_2 q_e^2} + \frac{t}{q_e} \quad (8)$$

Where q_t (mg g⁻¹) is the value of CR adsorbed at any time t (min), and k_1 (min⁻¹) and k_2 (g mg⁻¹ min⁻¹) are the rate constants of pseudo-first-order and pseudo-second-order rate constants, respectively. Linear regressions of the two models are shown in Fig. 4d and S2b (see ESI†), and the corresponding model constants and standard deviations as well as the coefficients of determination R^2 are summarized in Table 1. By comparison, the R^2 of the pseudo-second-order model is obviously superior than that of the pseudo-first-order model. Thus, the pseudo-second-order model can be fitted well with experimental data in this case. In addition, the $q_{e,cal}$ of pseudo-second-order model is also close to the value of $q_{e,exp}$, which further confirms that pseudo-second-order model can be more suitable to describe the adsorption process.

3.5 Adsorption mechanism

According to the work reported in literature,⁴⁰ the range of the free energy (ΔG^0) reflects indirectly the type of adsorption. The change of ΔG^0 of physisorption is between -20 and 0 kJ mol⁻¹. When ΔG^0 is ranging from -20 to -80 kJ mol⁻¹, the type of adsorption can be attributed to the physisorption together with chemisorption, and chemisorption is between -80 and -400 kJ mol⁻¹. The calculation formula of the changes in thermodynamic parameters of free energy (ΔG^0), enthalpy (ΔH^0), and entropy (ΔS^0) are calculated as the following [eqn (9) and (10)]:

$$\Delta G^0 = -RT \ln K_L \quad (9)$$

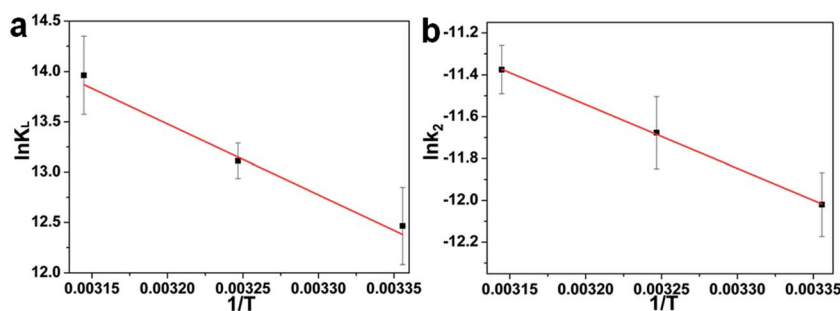
$$\ln K_L = \frac{\Delta S^0}{R} - \frac{\Delta H^0}{RT} \quad (10)$$

where R is the universal gas constant (8.314 J mol⁻¹ K⁻¹), K_L is the Langmuir equilibrium constant (L mol⁻¹) and T is the absolute temperature (K). ΔH^0 and ΔS^0 are determined from the slope and intercept of the van't Hoff plots of $\ln(K_L)$ versus $1/T$. In this case, the adsorption studies are also implemented at different temperatures (308 K and 318 K) for the initial CR concentration of 60, 200, 300, 400, 500, 600, 800, 1000 mg L⁻¹, respectively. The equilibrium adsorption isotherms of different temperature on CdO/CdFe₂O₄ HNNs and the corresponding linear dependence of C_e/q_e on C_e based on the Langmuir isotherm model are presented in Fig. S3a and b (see ESI†), respectively, and Table S3 (see ESI†) summarizes the linear correlation coefficients and standard deviations as well as the coefficients of determination R^2 . Subsequently, these results are plotted as error bars on the van't Hoff plots of $\ln(K_L)$ versus $1/T$ for the adsorption of CR onto CdO/CdFe₂O₄ HNNs in Fig. 5a. The calculations of ΔG^0 , ΔH^0 , and ΔS^0 are listed in Table 2. It



Table 1 Parameters and standard deviations (S.D.%) of kinetics model for adsorption of CR on CdO/CdFe₂O₄ HNNs at various initial concentrations

<i>C</i> ₀ (mg L ⁻¹)	Pseudo-first-order model				S.D. (%)	Pseudo-second-order model				
	Experimental <i>q</i> _e , (mg g ⁻¹)	Calculated <i>q</i> _e , (mg g ⁻¹)	<i>k</i> ₁ (g mg ⁻¹ min ⁻¹)	<i>R</i> ²		Experimental <i>q</i> _e , (mg g ⁻¹)	Calculated <i>q</i> _e , (mg g ⁻¹)	<i>k</i> ₂ (g mg ⁻¹ min ⁻¹)	<i>R</i> ²	S.D. (%)
60	150.0	112.1	5.44 × 10 ⁻²	0.646	71.3	150.0	155.5	1.53 × 10 ⁻³	0.998	1.05
100	248.7	157.9	1.59 × 10 ⁻²	0.889	27.1	248.7	250.0	2.70 × 10 ⁻⁴	0.992	1.48
150	374.2	289.6	1.29 × 10 ⁻²	0.942	15.5	374.2	377.3	9.74 × 10 ⁻⁵	0.984	1.41

**Fig. 5** Effect of temperature on the adsorption of CR on the porous CdO/CdFe₂O₄ HNNs (a) regressions of van't Hoff plot for thermodynamic parameters of $\ln K_L$ versus $1/T$ and (b) plots of $\ln k_2$ versus $1/T$. The error bar represents the standard deviations ($n = 3$).

can be seen that the value of ΔG^0 is from -30.88 to -36.91 kJ mol⁻¹, indicating that the adsorption of CR on 1D porous CdO/CdFe₂O₄ HNNs belongs to physisorption together with part of chemisorption. At the same time, the negative value of ΔG^0 further verify the adsorption process is spontaneous and feasibility. The values of ΔH^0 and ΔS^0 are calculated from the plot of $\ln K_L$ versus $1/T$ are 58.71 kJ mol⁻¹ and 299.94 J mol⁻¹ K⁻¹, respectively. The positive value of ΔH^0 indicates the process of adsorption is endothermic.^{41,42} Meanwhile, the positive value of ΔS^0 reflects the affinity of adsorbent for CR and indicates some structural changes in adsorbent and dye.

As mentioned above, the kinetic model for the adsorption of CR onto porous CdO/CdFe₂O₄ HNNs can be recognized as the pseudo-second-order model. Therefore, the activation energy of the adsorption process is calculated adopting the rate constants (k_2) of the pseudo-second-order model by using the Arrhenius equation as follows (eqn (11)):⁴³

$$\ln k_2 = \ln A - \frac{E_a}{RT} \quad (11)$$

where $\ln k_2$, E_a , A , R and T are the pseudo-second-order rate constant (g mg⁻¹ min⁻¹), activation energy (kJ mol⁻¹), pre-exponential factor (frequency factor), gas constant (8.314 J mol⁻¹ K⁻¹) and adsorption temperature (K), respectively. From the linear relationships and standard deviations between $\ln k_2$ and $1/T$ (Fig. 5b) for the process of adsorbing CR on the 1D HNNs, E_a is found to be 25.4 kJ mol⁻¹. The physisorption generally has activation energy ranging from 5–40 kJ mol⁻¹, while chemisorption has a larger activation energy (40–800 kJ mol⁻¹).^{42,44} As a consequence of ΔG^0 , ΔH^0 , and E_a , all values indicate that the adsorption process of CR on porous CdO/CdFe₂O₄ HNNs follows physisorption.

The removal of CR may be connected with the electrostatic attraction and surface complexation through hydrogen bonding between the adsorbent and CR molecules.^{19,44–48} As what has been discussed above, it would be reasonable to believe the existence of electrostatic attraction between CR and porous CdO/CdFe₂O₄ HNNs. In order to confirm the complexation effect, SEM and the FT-IR spectra of the as-prepared porous CdO/CdFe₂O₄ HNNs after test were also carried out. The SEM images of CR-loaded adsorbent (Fig. S4, see ESI[†]) show that the surface of the 1D HNNs is still rough, but no pores can be observed, which is a powerful proof that CR attach on the surface of adsorbent. FT-IR analysis further reveals the molecular and structural changes of CR and CR-loaded adsorbent. Fig. 6a shows the FT-IR spectrum of the CR-loaded adsorbent, in which the weak bond at 3448 cm⁻¹ can be attributed to the N–H stretching vibration. The bond at 1618 cm⁻¹ can be indexed to the C–C stretching vibration in benzene. The adsorption at 1448 and 1176 cm⁻¹ are due to the framework vibration of benzene rings. The adsorption at 1047 cm⁻¹ is the characteristic adsorption peak of S=O stretching vibration. Compared with the FT-IR spectrum of CR (Fig. 6b), there are two new peaks at 858.2 and 547.6 cm⁻¹, adsorptions at 858.2 cm⁻¹ is due to the Cd–O–H bending⁴⁹ and 547.6 cm⁻¹ represents Cd–O symmetrical stretching vibration.⁵⁰ It have been reported that the oxygen atom of the S=O group can be used as the hydrogen-bonding acceptor and produce intramolecular hydrogen bonding with the oxygen-containing functional group of alkali-activated carbon nanotubes.¹⁹ Here, S=O group of CR and the O group of CdO can form intramolecular hydrogen bonding and then produces Cd(OH)₂. Therefore, we can conclude that electrostatic attraction and surface complexation both contribute to the adsorption performance of CR on porous CdO/CdFe₂O₄



Table 2 Thermodynamics parameters for adsorption of CR on CdO/CdFe₂O₄ HNNs

Temperature (K)	ΔG_0 (kJ mol ⁻¹)	ΔH_0 (kJ mol ⁻¹)	ΔS_0 (J mol ⁻¹ K ⁻¹)	E_a (kJ mol ⁻¹)
298	-30.88			
308	-33.57	58.71	299.94	25.4
318	-36.91			

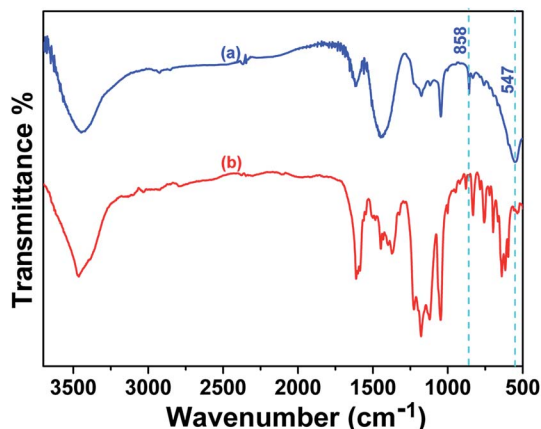


Fig. 6 FT-IR spectra of (a) CR-loaded adsorbent and (b) CR.

HNNs. In this case, the CR and adsorbent attract to each other under the effect of electrostatic attraction, and mesopores can provide a large number of active sites, which greatly promote the contact area. Additionally, the surface complexation *via* hydrogen bonding between the protonation hydroxyl of CdO/CdFe₂O₄ and sulfonate groups ($-\text{SO}_3^-$) of CR may be a vital factor for the adsorption.⁵¹ These results result in the excellent adsorption performance of the as-prepared 1D porous HNNs.

4. Conclusion

In conclusion, a new type of adsorbent based on 1D porous CdO/CdFe₂O₄ HNNs have been successfully prepared by a simple sacrificial template strategy. The solid Cd₂Fe(CN)₆ NNNs are first synthesized through a simple low-temperature hydrothermal process and then convert into the corresponding porous CdO/CdFe₂O₄ HNNs after thermal treatment. Benefiting the unique structural features, the resulting product was applied to remove the CR dye in aqueous solution, which exhibit outstanding adsorption performance with a maximum adsorption capacity of 1491 mg g⁻¹. The adsorption process of the CR on porous CdO/CdFe₂O₄ HNNs fits well with the pseudo-second-order rate equation and Langmuir adsorption model. We believe that this simple and scaled-up preparation strategy can be extended to synthesize other 1D porous nanostructures for various applications.

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References

- 1 L. Ai, H. Yue and J. Jiang, *Nanoscale*, 2012, **4**, 5401–5408.
- 2 H.-Y. Jing, T. Wen, C.-M. Fan, G.-Q. Gao, S.-L. Zhong and A.-W. Xu, *J. Mater. Chem. A*, 2014, **2**, 14563–14570.
- 3 L. Wu, Y. Liu, L. Zhang and L. Zhao, *Dalton Trans.*, 2014, **43**, 5393–5400.
- 4 W. Chen, W. Lu, Y. Yao and M. Xu, *Environ. Sci. Technol.*, 2007, **41**, 6240–6245.
- 5 G. Chen, *Sep. Purif. Technol.*, 2004, **38**, 11–41.
- 6 Y. Hu, X. Gao, L. Yu, Y. Wang, J. Ning, S. Xu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2013, **52**, 5636–5639.
- 7 X. Gao, H. B. Wu, L. Zheng, Y. Zhong, Y. Hu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2014, **53**, 5917–5921.
- 8 H. B. Wu, H. H. Hng and X. W. Lou, *Adv. Mater.*, 2012, **24**, 2567–2571.
- 9 A. Majcen-Le Marechal, Y. M. Slokar and T. Taufer, *Dyes Pigm.*, 1997, **33**, 281–298.
- 10 M. Zhou, X. Gao, Y. Hu, J. Chen and X. Hu, *Appl. Catal., B*, 2013, **138–139**, 1–8.
- 11 S. Chakraborty, M. K. Purkait, S. DasGupta, S. De and J. K. Basu, *Sep. Purif. Technol.*, 2003, **31**, 141–151.
- 12 L. Ai, C. Zhang and Z. Chen, *J. Hazard. Mater.*, 2011, **192**, 1515–1524.
- 13 Z. Yan, G. Li, L. Mu and S. Tao, *J. Mater. Chem.*, 2006, **16**, 1717–1725.
- 14 L. Ai, C. Zhang, F. Liao, Y. Wang, M. Li, L. Meng and J. Jiang, *J. Hazard. Mater.*, 2011, **198**, 282–290.
- 15 S. Babel and T. A. Kurniawan, *Chemosphere*, 2004, **54**, 951–967.
- 16 M. Ghaedi, A. Ansari, M. H. Habibi and A. R. Asghari, *J. Ind. Eng. Chem.*, 2014, **20**, 17–28.
- 17 Y. Yu, Y.-Y. Zhuang, Z.-H. Wang and M.-Q. Qiu, *Ind. Eng. Chem. Res.*, 2003, **42**, 6898–6903.
- 18 Y. Su, H. Cui, Q. Li, S. Gao and J. K. Shang, *Water Res.*, 2013, **47**, 5018–5026.
- 19 J. Ma, F. Yu, L. Zhou, L. Jin, M. Yang, J. Luan, Y. Tang, H. Fan, Z. Yuan and J. Chen, *ACS Appl. Mater. Interfaces*, 2012, **4**, 5749–5760.
- 20 J. B. Fei, Y. Cui, X. H. Yan, W. Qi, Y. Yang, K. W. Wang, Q. He and J. B. Li, *Adv. Mater.*, 2008, **20**, 452–456.
- 21 J. Fei, Y. Cui, J. Zhao, L. Gao, Y. Yang and J. Li, *J. Mater. Chem.*, 2011, **21**, 11742–11746.
- 22 W. Cai, J. Yu and M. Jaroniec, *J. Mater. Chem.*, 2010, **20**, 4587–4594.
- 23 B. Wang, H. Wu, L. Yu, R. Xu, T. T. Lim and X. W. Lou, *Adv. Mater.*, 2012, **24**, 1111–1116.
- 24 H. Xiao, Z. Ai and L. Zhang, *J. Phys. Chem. C*, 2009, **113**, 16625–16630.
- 25 F. Mou, J. Guan, H. Ma, L. Xu and W. Shi, *ACS Appl. Mater. Interfaces*, 2012, **4**, 3987–3993.
- 26 X. Y. Yu, R. X. Xu, C. Gao, T. Luo, Y. Jia, J. H. Liu and X. J. Huang, *ACS Appl. Mater. Interfaces*, 2012, **4**, 1954–1962.



- 27 X. Chen, F. Zhang, Q. Wang, X. Han, X. Li, J. Liu, H. Lin and F. Qu, *Dalton Trans.*, 2015, **44**, 3034–3042.
- 28 X. Sun, W. Luo, L. Chen, L. Zheng, C. Bao, P. Sun, N. Huang, Y. Sun, L. Fang and L. Wang, *RSC Adv.*, 2016, **6**, 2241–2251.
- 29 J. Wei, Y. Feng, Y. Liu and Y. Ding, *J. Mater. Chem. A*, 2015, **3**, 22300–22310.
- 30 A. Etogo, E. Hu, C. Zhou, Y. Zhong, Y. Hu and Z. Hong, *J. Mater. Chem. A*, 2015, **3**, 22413–22420.
- 31 R. Qiao, M. Mao, E. Hu, Y. Zhong, J. Ning and Y. Hu, *Inorg. Chem.*, 2015, **54**, 9033–9039.
- 32 A. Etogo, R. Liu, J. Ren, L. Qi, C. Zheng, J. Ning, Y. Zhong and Y. Hu, *J. Mater. Chem. A*, 2016, **4**, 13242–13250.
- 33 H.-Y. Zhu, R. Jiang, Y.-Q. Fu, R.-R. Li, J. Yao and S.-T. Jiang, *Appl. Surf. Sci.*, 2016, **369**, 1–10.
- 34 L. Wang, J. Li, Y. Wang, L. Zhao and Q. Jiang, *Chem. Eng. J.*, 2012, **181–182**, 72–79.
- 35 C. Lei, X. Zhu, Y. Le, B. Zhu, J. Yu and W. Ho, *RSC Adv.*, 2016, **6**, 10272–10279.
- 36 G. Li, Y. Sun, X. Li and Y. Liu, *RSC Adv.*, 2016, **6**, 11855–11862.
- 37 Z. Li, Y. Du, S. Zhang, Z. Chen, K. Yang, X. Lv and C. Zhu, *RSC Adv.*, 2016, **6**, 89699–89707.
- 38 Y. Zhang, S. Xu, Y. Luo, S. Pan, H. Ding and G. Li, *J. Mater. Chem.*, 2011, **21**, 3664.
- 39 J. Hu, Z. Song, L. Chen, H. Yang, J. Li and R. Richards, *J. Chem. Eng. Data*, 2010, **55**, 3742–3748.
- 40 A. Meng, J. Xing, Z. Li and Q. Li, *ACS Appl. Mater. Interfaces*, 2015, **7**, 27449–27457.
- 41 M. Kara, H. Yuzer, E. Sabah and M. S. Celik, *Water Res.*, 2003, **37**, 224–232.
- 42 Y. Yao, F. Xu, M. Chen, Z. Xu and Z. Zhu, *Bioresour. Technol.*, 2010, **101**, 3040–3046.
- 43 C. H. Wu, *J. Colloid Interface Sci.*, 2007, **311**, 338–346.
- 44 H. Nollet, M. Roels, P. Lutgen, P. Van der Meeren and W. Verstraete, *Chemosphere*, 2003, **53**, 655–665.
- 45 W. Q. Cai, J. G. Yu, B. Cheng, B. L. Su and M. Jaroniec, *J. Phys. Chem. C*, 2009, **113**, 14739–14746.
- 46 B. Cheng, Y. Le, W. Cai and J. Yu, *J. Hazard. Mater.*, 2011, **185**, 889–897.
- 47 F. Yu, J. Ma and Y. Wu, *J. Hazard. Mater.*, 2011, **192**, 1370–1379.
- 48 F. Yu, Y. Wu and J. Ma, *J. Hazard. Mater.*, 2012, **237–238**, 102–109.
- 49 V. R. Shinde, H. S. Shim, T. P. Gujar, H. J. Kim and W. B. Kim, *Adv. Mater.*, 2008, **20**, 1008–1012.
- 50 F. T. Thema, P. Beukes, A. Gurib-Fakim and M. Maaza, *J. Alloys Compd.*, 2015, **646**, 1043–1048.
- 51 C. C. Yu, X. P. Dong, L. M. Guo, J. T. Li, F. Qin, L. X. Zhang, J. L. Shi and D. S. Yan, *J. Phys. Chem. C*, 2008, **112**, 13378–13382.

