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Structures, stabilities and spectral properties of metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$)[†]

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The discovery of borospherene B_{40} marks the onset of a new class of boron fullerenes and it is of current interest in chemical physics and, in particular, boron chemistry. In this work, density functional theory (DFT) and time-dependent density functional theory (TD-DFT) calculations are carried out to study the structures, stabilities, photoelectron spectra, infrared spectra, Raman spectra and electronic absorption spectra of metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$). It is found that Cu, Ag and Au atoms can form stable exohedral metalloborospherenes $M\&B_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$) and endohedral metalloborospherenes $M@B_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$). In addition, the relative energies of these metalloborospherenes suggest that Cu, Ag and Au atoms favor the exohedral configuration. The Cu atom favors an exohedral geometry with the dopant face-capping a heptagon on the side surface ($\eta^7\text{-Cu}$), whereas Ag and Au atoms favor exohedral geometries with the dopant bonding a side boron atom of the hexagonal ring. The calculated spectra suggest that doping of metal atoms in borospherene B_{40} can change the spectral features since the extra metal atoms can modify the electronic structure of borospherene B_{40} . The addition of metal atoms can lead to more infrared and Raman active modes and redshift the electronic absorption spectra. The calculated results also show that metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$) have different and meaningful spectral features, insight into the spectral properties is important to understand the compounds and reveal their potential applications. These spectral features can be readily compared with future spectroscopy measurements and used as fingerprints to identify and distinguish the metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$).

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Introduction

Since the discovery of C_{60} ,¹ boron clusters have been received ample attention and previous works have shown that most boron clusters are planar or quasi-planar structures.²⁻⁹ In addition, stuffed fullerene-like boron carbide nanoclusters, boron sheets and different three-dimensional (bowl, tube, cage) boron clusters have been proposed based on theoretical calculations.¹⁰⁻²² Exhilaratingly, an all-boron fullerene B_{40}^- was produced using a laser vaporization supersonic source in 2014.²³ Photoelectron spectroscopy analysis confirmed the existence of B_{40}^- . The first all-boron fullerene with 40 atoms is named 'borospherene'. Soon after, the all-boron fullerene B_{39}^- was also produced *via* laser vaporization.²⁴ The first observation of borospherene has aroused interest in boron clusters and

their derivatives such as the dynamic behavior of B_{40} ,²⁵ hydrogen storage capacity of Ti-decorated B_{40} ,²⁶ manganese-centered and cobalt-centred tubular boron clusters (MnB_{16}^- and CoB_{16}^-),^{27,28} experimental and theoretical studies of B_n^- ($n = 27-29$),²⁹⁻³¹ structures and properties of borospherenes (B_{42}^+ , $B_{44}^{0/-}$, and B_{46}),³²⁻³⁵ structures of boron clusters B_n ($n = 30$ to 51) and exohedral complexation in $C_6H_6CrB_{40}$,^{36,37} CO_2 capture, storage and separation by borospherene B_{40} ,³⁸ and spectral properties of borospherenes $B_{40}^{0/-}$.^{39,40}

Since the discovery of fullerene C_{60} ,¹ researchers have reported the development of several fullerene derivatives based on the functionalization of the C_{60} aiming to produce new properties or improve their existing properties. The important studies involve the use of C_{60} derivatives in biological systems,^{41,42} solar cells^{43,44} and superconductors.⁴⁵ Endofullerenes based on C_{60} have attracted much attention due their potential applications in superconductivity and materials science.⁴⁶⁻⁴⁹ Similarly, can metalloborospherenes also show new properties? Several theoretical studies have been reported to investigate the structures and electronic properties of metalloborospherenes.⁵⁰⁻⁵⁷ Bai *et al.* reported the metalloborospherenes MB_{40} ($M = Ca, Mg, Sr, \text{ and } Be$), they found that both Ca and Sr with bigger atomic radii exhibit endohedral structures as the global minima, whereas Be and Mg with smaller atomic radii favor the exohedral geometries with the

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dopant face-capping a heptagon on the side surface (η^7 -M). Their studies suggest that the atomic radii of metal dopants play a key factor in determining stability of the metalloborospherenes. Jin *et al.* reported the metalloborospherenes MB_{40} ($M = Sc, Y,$ and La). They found that $Sc, Y,$ and La atoms all favor the off-center location within the B_{40} cage. However, in another study of metalloborospherenes MB_{40} ($M = Li, Na, Ba,$ and K), the results show that K favors the η^7 -K exohedral location whereas Na favors the endohedral off-center location, even though the size of Na (1.91 \AA) is smaller than that of K (2.34 \AA). So the atomic radius of metal dopant is not the only factor in determining stability of the metalloborospherenes. Therefore, it will be useful to determine additional factors that can be used to identify the preference of endohedral or exohedral doping for the metal atoms. It is necessary to examine all possible binding configurations of other metal atoms with the B_{40} cage.

Ag and Au have often been used as electrodes of single-molecule device, such as the electrical transport properties of single-molecule devices $Au-B_{40}-Au$ and $Au-Sr@B_{40}-Au$ were studied by using density functional theory and non-equilibrium Green's function method.⁵⁸⁻⁶⁰ It is necessary to investigate the structures and stabilities of metalloborospherenes $MB_{40}^{0/-}$ ($M = Ag, Au$). On the one hand, it can enrich the research of metalloborospherenes, on the other hand, it also can provide theoretical guidance for borospherene or metalloborospherene molecular devices. In this work, we provide a theoretical study on the structures, stabilities, photoelectron spectra, infrared, Raman and electronic absorption spectra of metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag,$ and Au). Our works can provide valuable results to assist further experimental identifications on the metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag,$ and Au), and also may provide theoretical guidance for the applications and synthesis of them in the future.

Computational methods

To gain relative stabilities between the endohedral and exohedral structures of metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag,$ and Au), we divided into two steps. First, we performed a scan of potential energy by using the dopant as a probe to move along the path connecting the B_{40} center toward the outside but keeping the B_{40} cage intact. Considering the molecular symmetry (D_{2d}), six target paths were selected. (1) L_y : connecting the B_{40} center and a heptagon center (y axis); (2) L_z : connecting the B_{40} center and a hexagon center (z axis); (3) L_{xy} : angle bisector of angle between x axis and y axis; (4) L_{yz} : angle bisector of angle between y axis and z axis; (5) L_{xz} : angle bisector of angle between x axis and z axis; (6) L_{xyz} : intersection line of three planes determined by x axis and L_{yz} , z axis and L_{xz} , y and L_{xz} , respectively. Structure of D_{2d} B_{40} indicates that there exist six kinds of boron atoms with different chemical environment (confirmed by the NBO charge analysis and calculated ^{11}B NMR spectrum^{61,62}), two of which belong to hexagon and four to heptagon as shown by different numbers in Fig. 1. Four paths containing L_{xz} , L_{xy} , L_{yz} and L_{xyz} ensure the dopant atop six different boron atoms. This method is a quick and advisable computational indicator to demonstrate the relative stabilities

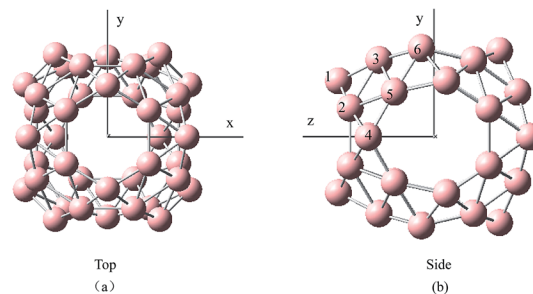


Fig. 1 Structures of $B_{40}^{0/-}$ observed along the (a) z direction and (b) x direction. Six numbers denote the six kinds of boron atoms with different chemical environment.

between the endohedral and exohedral configurations of metallofullerenes and metalloborospherenes.^{52,63} In the last step, geometry optimizations and frequency calculations were performed containing local minima in the six target paths. The potential energy scan was performed using the PBE0 functional with 6-31G basis set for boron and a double- ζ basis set (LanL2DZ) with the effective core potential for metals. Geometry optimizations and harmonic frequency analysis were carried out using the PBE0 functional with 6-311+G* basis set for boron and a double- ζ basis set (LanL2DZ) with the effective core potential for metals. These optimized structures were then used in the calculations of photoelectron spectra and electronic absorption spectra based on the time-dependent DFT formalism.⁶⁴ Adiabatic detachment energy (ADE) was calculated as the energy difference of the anion and neutral metalloborospherenes at their respective optimized geometries. The first vertical detachment energy (VDE) was calculated as the energy difference between the anion ground state and the neutral at the optimized anion geometry. The higher vertical detachment energies (VDEs) were approximated *via* Δ SCF-time-dependent density functional theory (Δ SCF-TD-DFT).^{23,24,64} All computations were carried out using the Gaussian09 software package.⁶⁵

Results and discussion

Structures

Fig. 2–7 are results of potential energy scan by moving the metal dopant from the cage center through different paths while keeping the cage intact. The open location of each curve is just cage surface in the presence of boron atoms. The results indicate that these potential energy curves exhibit two local minima: one corresponds to the endohedral structure, whereas another corresponds to the exohedral structure. We also notice that the exohedral minimum is lower in energy than the endohedral minimum. For endohedral $Ag@B_{40}$, $Ag@B_{40}^-$ and $Au@B_{40}$, when the dopant is at the center of cage, endohedral structure is local minimum. However, for both $Cu@B_{40}$ and $Cu@B_{40}^-$, endohedral structure which the dopant is at the off-center location within the cage is local minimum. In addition, endohedral $Au@B_{40}^-$ which the dopant is at the slightly off-center location within the cage is local minimum.



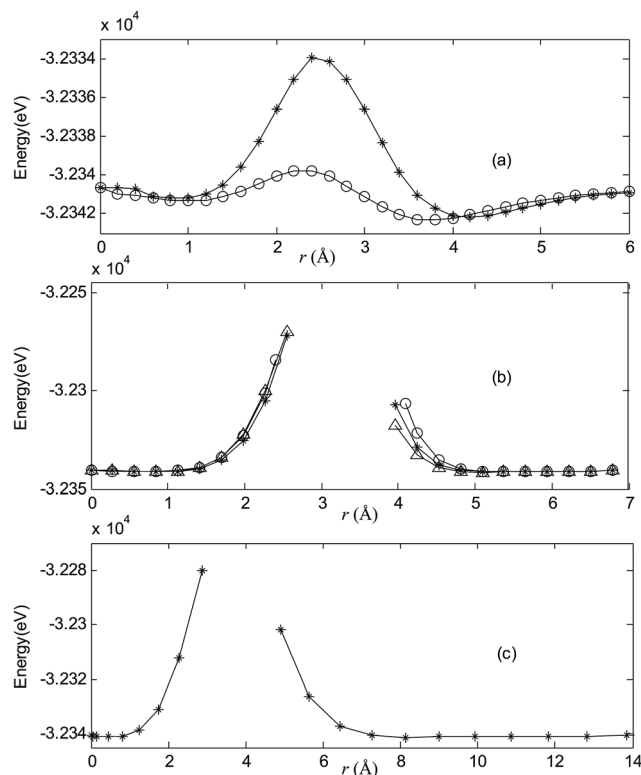


Fig. 2 Potential energy scan by moving the metal dopant Cu from the cage center through different paths while keeping the cage intact for CuB_{40} . r represents the distance between the dopant Cu and the cage center. (a) Asterisk represents the path L_z , circle represents the path L_y , (b) triangle represents the path L_{yz} , asterisk represents the path L_{xz} , circle represents the path L_{xz} , (c) the path L_{xyz} .

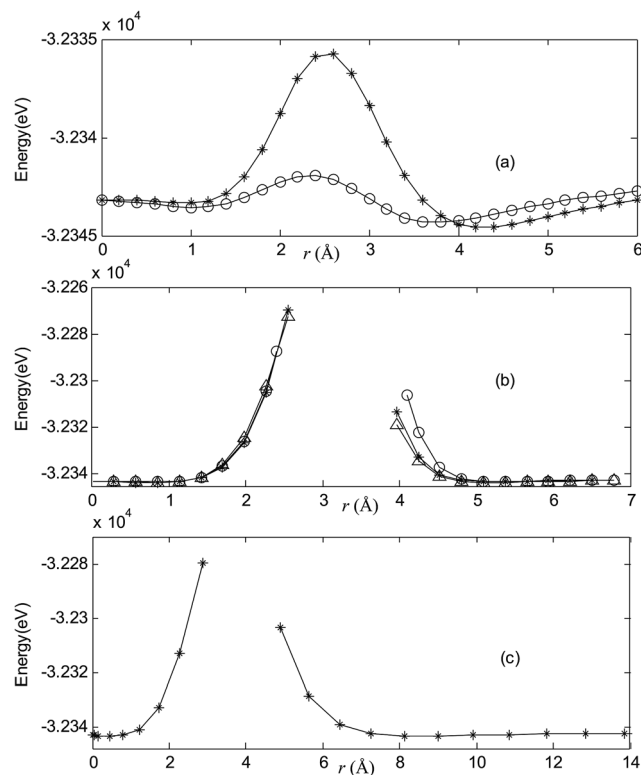


Fig. 3 Potential energy scan by moving the metal dopant Cu from the cage center through different paths while keeping the cage intact for CuB_{40}^- . r represents the distance between the dopant Cu and the cage center. (a) Asterisk represents the path L_z , circle represents the path L_y , (b) triangle represents the path L_{yz} , asterisk represents the path L_{xz} , circle represents the path L_{xz} , (c) the path L_{xyz} .

For all metalloborosphenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$), as illustrated in Fig. 2–7, the energy rapidly increases near the cage surface and exhibits a maximum at the cage surface. In addition, the energy rapidly increases near the hexagonal and heptagonal rings and exhibits a maximum at the center of the hexagonal and heptagonal ring, respectively. The energy barriers for the Cu, Ag, and Au dopants are 1.54 eV (CuB_{40}), 1.61 eV (CuB_{40}^-), 7.72 eV (AgB_{40}), 7.92 eV (AgB_{40}^-), 7.55 eV (AuB_{40}), and 7.29 eV (AuB_{40}^-) for their moving “in” through the heptagon ring, respectively. The energy barriers for the $\text{CuB}_{40}^{0/-}$, $\text{AgB}_{40}^{0/-}$, and $\text{AuB}_{40}^{0/-}$ dopants are 7.18 eV (CuB_{40}), 7.70 eV (CuB_{40}^-), 19.09 eV (AgB_{40}), 20.27 eV (AgB_{40}^-), 20.98 eV (AuB_{40}), 21.79 eV (AuB_{40}^-) for their moving “in” through the hexagon ring, respectively. The larger the ionic radius, the higher is the energy barrier. The large barrier height found for Ag and Au indicates that the atom cannot escape from B_{40} once encapsulated. In addition, for a dopant, the energy barrier is smaller for the heptagon ring, indicating stronger interaction for the hexagon ring. The results can be understood by comparing the size of the hexagonal and heptagonal ring.⁶³ The hexagonal ring radius is smaller compared with the heptagonal ring radius, but smaller radius leads to stronger interaction to pass through the hexagonal ring with closer touching the electrons responsible for the B–B bond. The heptagonal ring radius is already larger

than the hexagonal ring radius, where the energy barrier is smaller for heptagonal ring.

The above results indicate that the exohedral minimum is lower in energy than the endohedral minimum. To gain the further understanding of relative stabilities between the endohedral and exohedral structures of metalloborosphenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$), geometry optimizations and frequency calculations were performed for the two local minima in the six paths. Optimized structures of metalloborosphenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$) are depicted in Fig. S1–S3 (ESI†). Relative energy values of neutral and corresponding anionic metalloborosphenes $\text{MB}_{40}^{0/-}$ indicate that the exohedral minimum is lower in energy than the endohedral minimum. The results also reveal that Cu, Ag and Au atoms favor the exohedral configuration. For neutral and anionic metalloborosphenes $\text{CuB}_{40}^{0/-}$, the isomer (see Fig. S1(a)†) with Cu capping a heptagon ($\eta^7\text{-Cu}$) is lower in energy than another exohedral and endohedral isomers. The results are same with the found that MB_{40} ($M = \text{Be}, \text{Mg}, \text{Li}, \text{and K}$) favor the exohedral geometries with the dopant face-capping a heptagon on the side surface ($\eta^7\text{-M}$). For neutral and anionic metalloborosphenes $\text{MB}_{40}^{0/-}$ ($M = \text{Ag and Au}$), the isomer (see Fig. S2(a) or S3(a)†) with dopant bonding the side boron atom of the hexagonal ring is lower in energy than another exohedral and endohedral isomers. This isomer with C_s symmetry is predicted after the



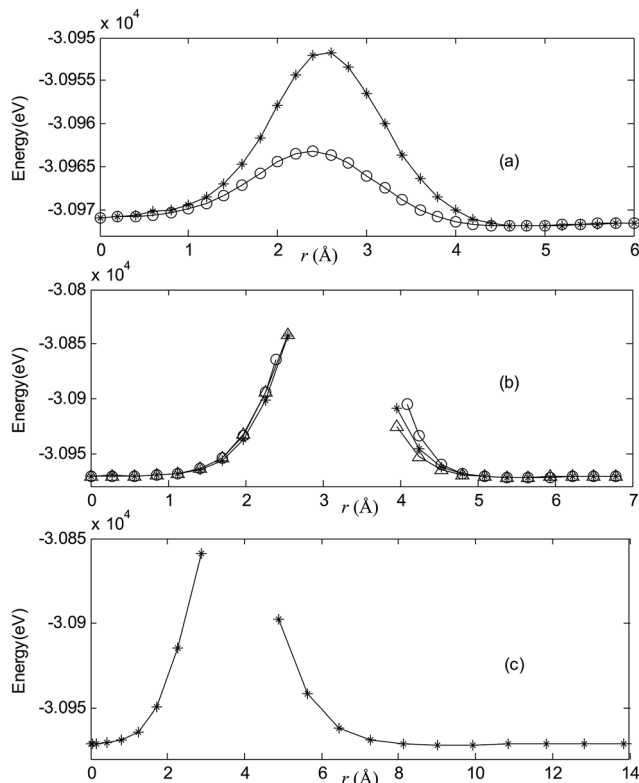


Fig. 4 Potential energy scan by moving the metal dopant Ag from the cage center through different paths while keeping the cage intact for AgB_{40} . r represents the distance between the dopant Ag and the cage center. (a) Asterisk represents the path L_z , circle represents the path L_y , (b) triangle represents the path L_{yz} , asterisk represents the path L_{xz} , circle represents the path L_{xy} , (c) the path L_{xyz} .

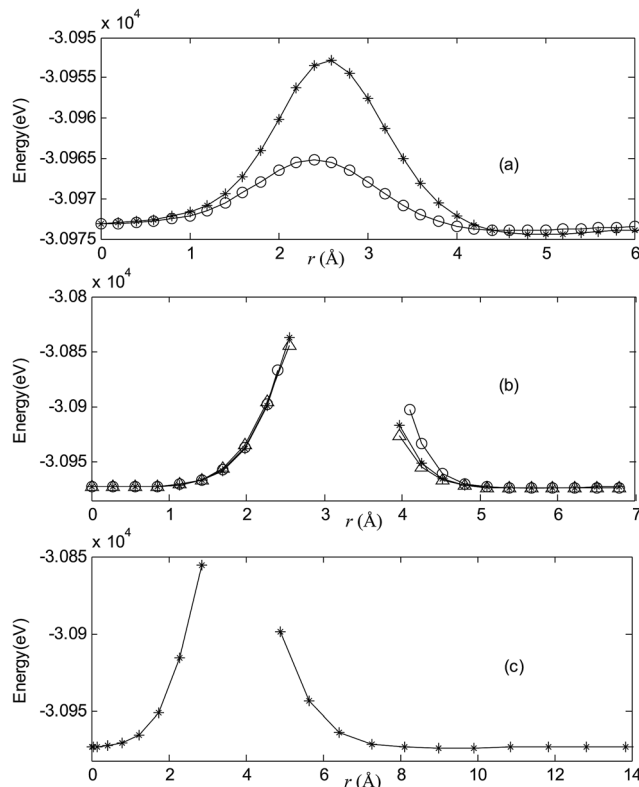


Fig. 5 Potential energy scan by moving the metal dopant Ag from the cage center through different paths while keeping the cage intact for AgB_{40}^- . r represents the distance between the dopant Ag and the cage center. (a) Asterisk represents the path L_z , circle represents the path L_y , (b) triangle represents the path L_{yz} , asterisk represents the path L_{xz} , circle represents the path L_{xy} .

relaxation in the path L_{yz} . The result further proves the research results that single-molecule device $\text{Au-B}_{40}\text{-Au}$ which electrode is connected to the hexagonal ring is the most stable configuration.⁶⁰ For most stable endohedral metalloborospherenes $\text{M@B}_{40}^{0/-}$, the calculated results indicate that Cu atom in Cu@B_{40}^- and Cu@B_{40} are off the molecular center and near the heptagonal ring. However, Ag atom in Ag@B_{40}^- and Au atom in Au@B_{40} are at the molecular center. In addition, Au atom in Au@B_{40}^- is slightly off the molecular center. It's worth noting that configuration of Au@B_{40}^- has obvious changes and symmetry of Au@B_{40}^- reduces to C_1 . Frequency calculations confirm the stabilities of these most stable endohedral and exohedral metalloborospherenes by showing no imaginary frequencies. We will focus on the most stable endohedral and exohedral metalloborospherenes. The most stable endohedral and exohedral metalloborospherenes are depicted in Fig. 8–10, ground-state parameters are summarized in Table 1. As shown in Table 1, HOMO–LUMO energy gaps of metalloborospherenes MB_{40}^- (with closed-shell electronic structure) are smaller than HOMO–LUMO energy gap of borospherene B_{40} (3.13 eV).²³ It indicates that the addition of metal atom (Cu, Ag, or Au) can reduce the HOMO–LUMO energy gap and improve the chemical activity of borospherene B_{40} . In addition, dipole moments of $\text{Ag@B}_{40}^{0/-}$ and Au@B_{40} are zero among all the

metalloborospherenes because of the highly symmetric structures (D_{2d}), this indicates that they do not render far-infrared pure rotation spectrum. For each neutral metalloborospherene, the addition of metal with odd electrons introduces spin density into the boron framework (see Fig. S4†). For metalloborospherenes Cu@B_{40} and Cu@B_{40}^- , the spin components are mainly distributed in the surface of the cage. For metalloborospherenes Ag@B_{40} and Au@B_{40} , the spin components are mainly localized on metal and boron atoms in close proximity to metal. For metalloborospherenes Ag@B_{40} and Au@B_{40} , the spin components are mainly localized on metal (or cage centers). Especially, for metalloborospherenes AgB_{40} and AuB_{40} , the up-spin components are mainly contributed by the metal. These characteristics are expected to lead to interesting magnetic properties and potential application in metalloborospherene molecular devices.

Photoelectron spectra

Photoelectron spectroscopy is powerful experimental technique to probe the electronic structure of cluster. It can be viewed as an electronic fingerprint for the underlying cluster. Photoelectron spectroscopy in combination with theoretical calculations has been used to understand and identify the structures of size-selected boron clusters.^{23,24} To facilitate future



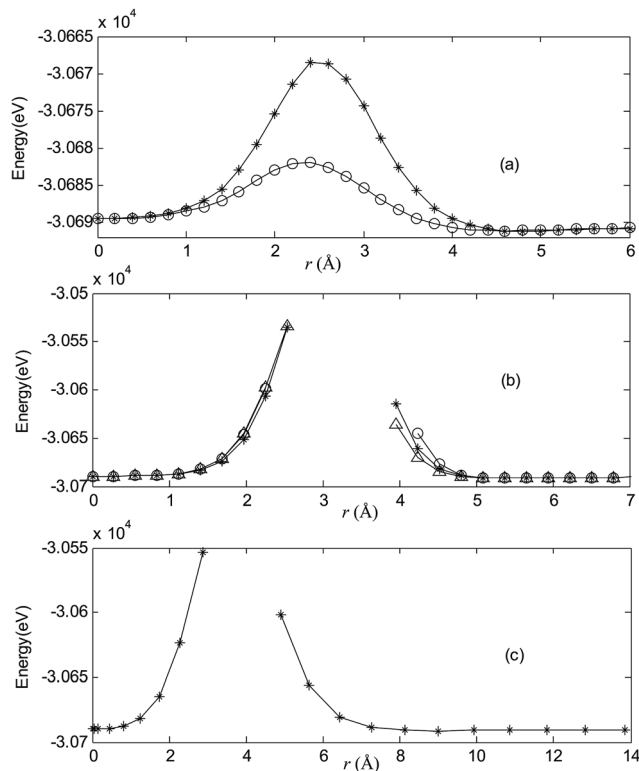


Fig. 6 Potential energy scan by moving the metal dopant Au from the cage center through different paths while keeping the cage intact for AuB_{40}^- . r represents the distance between the dopant Au and the cage center. (a) Asterisk represents the path L_z , circle represents the path L_y , (b) triangle represents the path L_{yz} , asterisk represents the path L_{xz} , circle represents the path L_{xyz} .

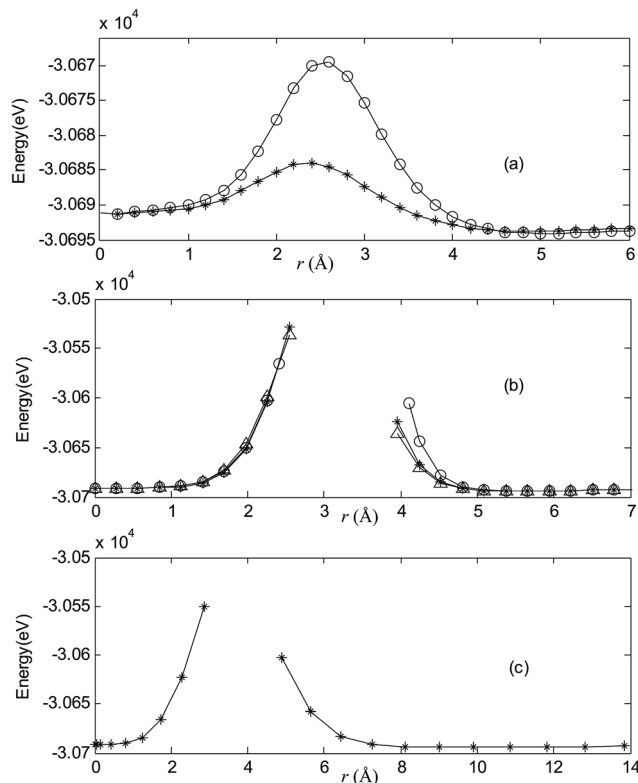


Fig. 7 Potential energy scan by moving the metal dopant Au from the cage center through different paths while keeping the cage intact for AuB_{40}^- . r represents the distance between the dopant Au and the cage center. (a) Asterisk represents the path L_z , circle represents the path L_y , (b) triangle represents the path L_{yz} , asterisk represents the path L_{xz} , circle represents the path L_{xyz} .

identifications of MB_{40}^- ($M = \text{Cu, Ag, and Au}$), the adiabatic detachment energies (ADEs) for metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$) were calculated at the PBE0 level, then we calculated the vertical detachment energies (VDEs) and simulated the photoelectron spectra for metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$), using the time-dependent DFT (TD-DFT) method.^{23,24,64} Adiabatic detachment energy of MB_{40}^- ($M = \text{Cu, Ag, and Au}$) represents the electron affinity (EA) of corresponding neutral metalloborospherenes MB_{40} ($M = \text{Cu, Ag, and Au}$). The larger EA can lead to the stronger probability of capturing an electron, *i.e.*, the neutral MB_{40} with larger EA is easier to capture an electron. Metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$) give the ground-state ADEs of 2.38(Cu@B_{40}^-), 2.53(Cu@B_{40}^-), 2.87(Ag@B_{40}^-), 2.53(Ag@B_{40}^-), 3.34(Au@B_{40}^-) and 2.94(Au@B_{40}^-) eV, respectively. Among the metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$), Au@B_{40}^- has the largest ADE (3.34 eV), which is larger than the ADE (2.29 eV)²³ of cage B_{40}^- and less than the ADE (3.51 eV)²³ of quasi-planar B_{40}^- . The calculated results indicate that ground-state ADEs of metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$) are larger than the ADE (2.29 eV)²³ of cage B_{40}^- and less than the ADE (3.51 eV)²³ of quasi-planar B_{40}^- , it suggests that doping of metal atom in B_{40} can increase the ADE.

Fig. 11 presents the photoelectron spectra (calculated VDEs were tabulated in Table S1†) of metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$). The predicted photoelectron spectra show

that Cu@B_{40}^- has the lowest first vertical detachment energy (VDE) and Ag@B_{40}^- has the largest energy gap (about 1.36 eV) between the first and second bands. The first several bands of photoelectron spectra were used to identify boron clusters,^{23,24} so we will focus on the bands at the low binding energy side. The first peaks of these photoelectron spectra come from the calculated ground-state VDEs of Cu@B_{40}^- , Cu@B_{40}^- , Ag@B_{40}^- , Ag@B_{40}^- , Au@B_{40}^- and Au@B_{40}^- at 2.46, 2.64, 2.98, 2.63, 3.46 and 3.60 eV, respectively. The calculated ground-state VDEs of metalloborospherenes MB_{40}^- ($M = \text{Cu, Ag, and Au}$) originate from the detachment of the electron from the molecular orbital (HOMO). The second peak of Cu@B_{40}^- comes from the second and third calculated VDEs at 3.22 and 3.25 eV which originate from detaching the electrons from HOMO-1 and HOMO-2. The second peaks of Cu@B_{40}^- come from the second and third calculated VDEs at 3.49 and 3.50 eV. The second peak of Ag@B_{40}^- comes from the second to fourth calculated VDEs at 4.34, 4.37, and 4.39 eV, respectively. The second to fourth calculated VDEs of metalloborospherene Ag@B_{40}^- originate from detaching the electrons from HOMO-1 to HOMO-3. The second peak of Ag@B_{40}^- comes from the second and third calculated VDEs, interestingly, the second and third calculated VDEs of metalloborospherene Ag@B_{40}^- are overlapped. The second peak of Au@B_{40}^- comes from the second to fourth calculated VDEs at 4.45, 4.52, and 4.54 eV, respectively. The



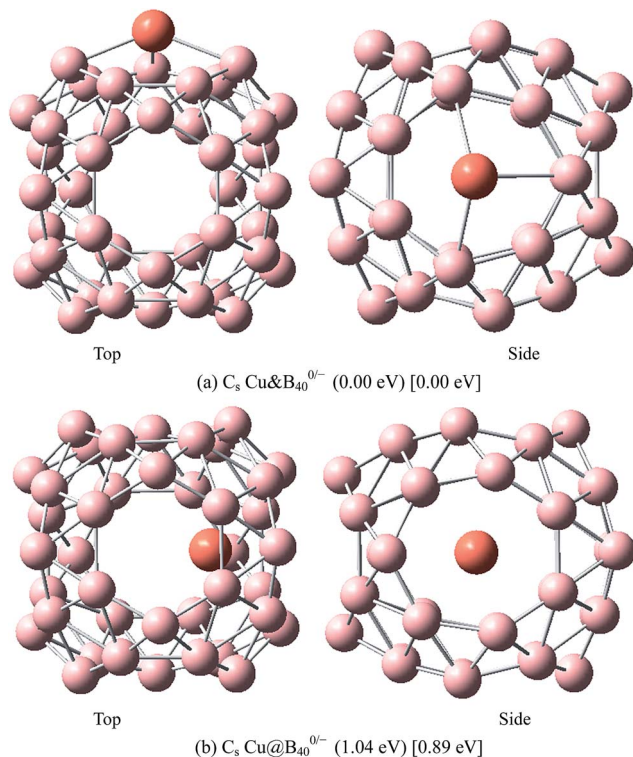


Fig. 8 The most stable endohedral and exohedral metalloborosphenes $\text{CuB}_{40}^{0/-}$. The round brackets denote the relative energies of neutral metalloborosphenes CuB_{40} (the energy of exohedral $\text{Cu}\&\text{B}_{40}$ is set to be zero) and the square brackets denote the relative energies of anionic metalloborosphenes CuB_{40}^- (the energy of exohedral $\text{Cu}\&\text{B}_{40}^-$ is set to be zero).

second to fourth calculated VDEs of metalloborosphere $\text{Au}\&\text{B}_{40}^-$ originate from detaching the electrons from HOMO–1 to HOMO–3. The second and third peaks of $\text{Au}@B_{40}^-$ come from the second and third calculated VDEs at 4.17 and 4.38 eV, respectively. Fig. 11 indicates that metalloborosphenes MB_{40}^- ($M = \text{Cu}, \text{Ag}, \text{and Au}$) have different spectral features, especially the endohedral and corresponding exohedral metalloborosphenes have different spectral bands at the low binding energy side. These features can be used to distinguish the metalloborosphenes MB_{40}^- ($M = \text{Cu}, \text{Ag}, \text{and Au}$).

Fig. 11(a–f) and photoelectron spectra of borospherene B_{40}^- (see Fig. 11(g))²³ indicate that the addition of metal atom modifies the photoelectron spectra of borospherene B_{40}^- since the addition of extra atom modifies the electronic structure. The calculated results indicate that doping of metal atom in borospherene can increase the first VDE compared to the first VDE of B_{40}^- . The predicted photoelectron spectra in Fig. 11 provide important information for the identification of metalloborosphenes MB_{40}^- ($M = \text{Cu}, \text{Ag}, \text{and Au}$). It is worth to note that the structures of atomic clusters cannot directly be identified by common analytical experimental methods, but they can indirectly be determined by using combined theoretical and experimental studies. As the discovery of B_{40} , if the photoelectron spectra of metalloborosphenes MB_{40}^- ($M = \text{Cu}, \text{Ag}, \text{and Au}$) are obtained in experiments, these calculated characteristic

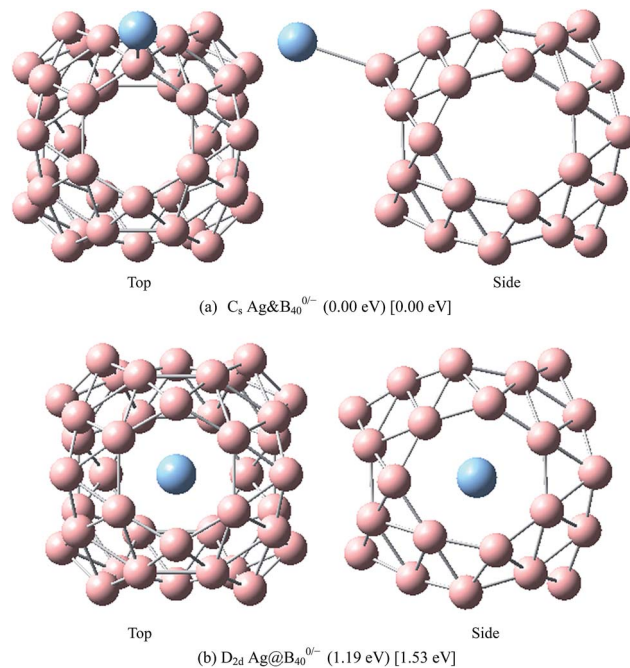


Fig. 9 The most stable endohedral and exohedral metalloborosphenes $\text{AgB}_{40}^{0/-}$. The round brackets denote the relative energies of neutral metalloborosphenes AgB_{40} (the energy of exohedral $\text{Ag}\&\text{B}_{40}$ is set to be zero) and the square brackets denote the relative energies of anionic metalloborosphenes AgB_{40}^- (the energy of exohedral $\text{Ag}\&\text{B}_{40}^-$ is set to be zero).

bands may be used as theoretical basis for the identification of metalloborosphenes MB_{40}^- ($M = \text{Cu}, \text{Ag}, \text{and Au}$).

Infrared spectra

Infrared spectra of metalloborosphenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$) are given in Fig. 12 (spectral data were tabulated in Tables S2–S13[†]), these infrared peaks distribute in three regions: low frequency region (from 0 cm^{-1} to 500 cm^{-1}), middle frequency region (from 500 cm^{-1} to 1000 cm^{-1}) and high frequency region (from 1000 cm^{-1} to 1400 cm^{-1}), and the main strong infrared peaks are located in the high frequency region (from 1000 cm^{-1} to 1400 cm^{-1}). The highest frequency is 1323 cm^{-1} for $\text{Cu}\&\text{B}_{40}$, 1319 cm^{-1} for $\text{Cu}\&\text{B}_{40}^-$, 1314 cm^{-1} for $\text{Cu}@B_{40}$, 1305 cm^{-1} for $\text{Cu}@B_{40}^-$, 1320 cm^{-1} for $\text{Ag}\&\text{B}_{40}$, 1314 cm^{-1} for $\text{Ag}\&\text{B}_{40}^-$, 1307 cm^{-1} for $\text{Ag}@B_{40}$, 1299 cm^{-1} for $\text{Ag}@B_{40}^-$, 1320 cm^{-1} for $\text{Au}\&\text{B}_{40}$, 1316 cm^{-1} for $\text{Au}\&\text{B}_{40}^-$, 1304 cm^{-1} for $\text{Au}@B_{40}$, and 1306 cm^{-1} for $\text{Au}@B_{40}^-$. These vibrational modes are formed by stretching vibrations of boron atoms. In addition, they are close to the highest vibrational frequency of $D_{2d} \text{B}_{40}$ (1327 cm^{-1}). The lowest vibrational frequency is 116 cm^{-1} for $\text{Cu}\&\text{B}_{40}$, 107 cm^{-1} for $\text{Cu}\&\text{B}_{40}^-$, 89 cm^{-1} for $\text{Cu}@B_{40}$, 92 cm^{-1} for $\text{Cu}@B_{40}^-$, 25 cm^{-1} for $\text{Ag}\&\text{B}_{40}$, 36 cm^{-1} for $\text{Ag}\&\text{B}_{40}^-$, 47 cm^{-1} for $\text{Ag}@B_{40}$, 52 cm^{-1} for $\text{Ag}@B_{40}^-$, 34 cm^{-1} for $\text{Au}\&\text{B}_{40}$, 44 cm^{-1} for $\text{Au}\&\text{B}_{40}^-$, 25 cm^{-1} for $\text{Au}@B_{40}$, and 67 cm^{-1} for $\text{Au}@B_{40}^-$. These vibrational modes are formed by vibrations of metal atoms. Unlike the highest frequencies of metalloborosphenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$), these lowest vibrational frequencies of



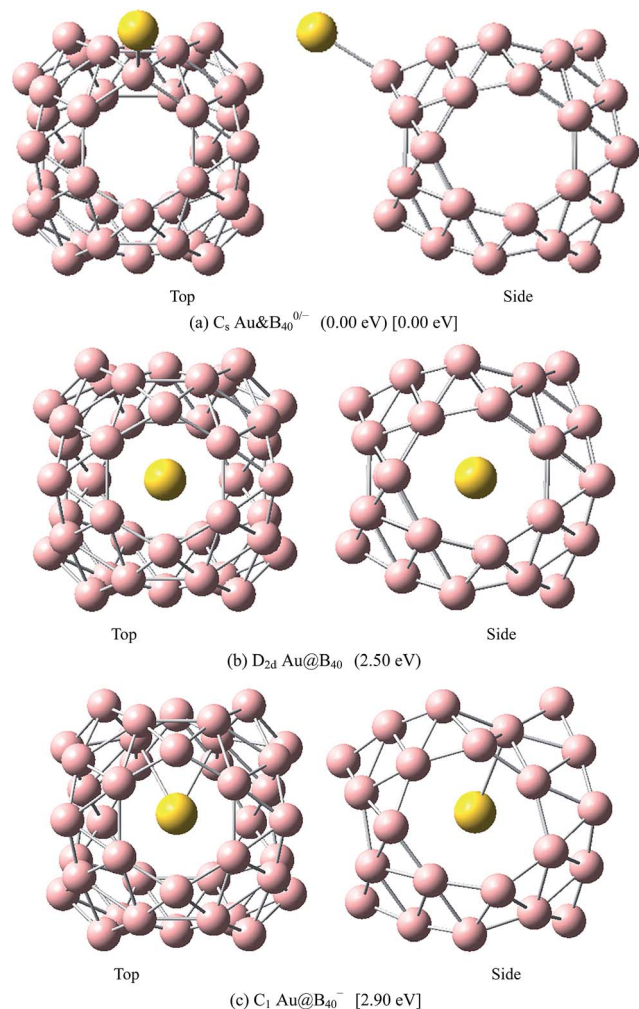


Fig. 10 The most stable endohedral and exohedral metalloborospherenes AuB₄₀⁰⁻. The round brackets denote the relative energies of neutral metalloborospherenes AuB₄₀ (the energy of exohedral Au&B₄₀ is set to be zero) and the square brackets denote the relative energies of anionic metalloborospherenes AuB₄₀⁻ (the energy of exohedral Au&B₄₀⁻ is set to be zero).

metalloborospherenes MB₄₀⁰⁻ (M = Cu, Ag, and Au) are much lower than the lowest vibrational frequency of borospherene B₄₀ (170 cm⁻¹).

The sharpest peaks of Cu&B₄₀, Cu&B₄₀⁻, Cu@B₄₀, Cu@B₄₀⁻, Ag&B₄₀, Ag&B₄₀⁻, Ag@B₄₀, Ag@B₄₀⁻, Au&B₄₀, Au&B₄₀⁻, Au@B₄₀ and Au@B₄₀⁻ are at 1275, 1265, 1265, 1266, 1270, 1262, 1278, 1271, 1271, 1268, 1247 and 1217 cm⁻¹, respectively. These vibrational modes are formed by stretching vibrations of boron atoms. Note that the most sharpest peaks of metalloborospherenes MB₄₀⁰⁻ (M = Cu, Ag, and Au) are located at about 1270 cm⁻¹ and almost same with the location (at 1274 cm⁻¹) of sharpest peak for borospherene B₄₀.⁴⁰ Owing to the high symmetry (D_{2d}) of Ag@B₄₀, Ag@B₄₀⁻, and Au@B₄₀, there are doubly degenerate vibrational modes. However, the other structures have only non-degenerate vibrational modes. Like the infrared spectra of borospherene B₄₀,^{39,40} some vibrational modes of endohedral Ag@B₄₀⁰⁻ and Au@B₄₀ are infrared

inactive. However, all the vibrational modes of other structures are infrared active modes, it suggest that doping of metal atom can add the spectra active modes.

It's worth noting that Fig. 12(e and i) show that exohedral Ag&B₄₀ and Au&B₄₀ have similar infrared spectra, and Fig. 12(f and j) show that exohedral Ag&B₄₀⁻ and Au&B₄₀⁻ have similar infrared spectra. In addition, Fig. 12(g and h) show that endohedral Ag@B₄₀ and Ag@B₄₀⁻ have similar infrared spectra, but the main peaks of anionic Ag@B₄₀⁻ are redshifted somewhat since the extra electron modifies the electronic structures. It's worth noting that the infrared spectra of endohedral Ag@B₄₀⁰⁻ are quite similar to that of dianion B₄₀⁻ (see Fig. 12(n)).⁴⁰ However, Fig. 12(k) shows that the infrared spectra of endohedral Au@B₄₀ is similar to that of B₄₀ (see Fig. 12(m)). Except for Ag@B₄₀⁰⁻ and Au@B₄₀, infrared spectra of other metalloborospherenes are different from that of B₄₀⁰⁻,⁴⁰ the metal dopant in B₄₀ changes the IR spectra of B₄₀ such as some weakened vibrational modes and some enhanced characteristic peaks. These spectral characteristics suggest that location of dopant and charge of metalloborospherene play an important role in infrared spectra. The predicted infrared spectra also provide some information for the identification of metalloborospherenes MB₄₀⁰⁻ (M = Cu, Ag, and Au), these different characteristic peaks also provide a theoretical basis for the identification and confirmation of metalloborospherenes MB₄₀⁰⁻ (M = Cu, Ag, and Au).

Raman spectra

Fig. 13 depicts the Raman spectra of metalloborospherenes MB₄₀⁰⁻ (M = Cu, Ag, and Au), the sharpest peaks of Cu&B₄₀, Cu&B₄₀⁻, Cu@B₄₀, Cu@B₄₀⁻, Ag&B₄₀, Ag&B₄₀⁻, Ag@B₄₀, Ag@B₄₀⁻, Au&B₄₀, Au&B₄₀⁻, Au@B₄₀ and Au@B₄₀⁻ are at 435, 1265, 1238, 654, 685, 672, 1229, 775, 1320, 737, 1304 and 1306 cm⁻¹, respectively. Unlike the infrared spectra of MB₄₀⁰⁻ (M = Cu, Ag, and Au), the sharpest Raman peaks of metalloborospherenes MB₄₄⁰⁻ (M = Li, Na, and K) are at middle or high frequency region. Among the Raman active modes, the typical radial breathing modes of Cu&B₄₀, Cu&B₄₀⁻, Cu@B₄₀, Cu@B₄₀⁻, Ag&B₄₀, Ag&B₄₀⁻, Ag@B₄₀, Ag@B₄₀⁻, Au&B₄₀, Au&B₄₀⁻ and Au@B₄₀ are at 166, 145, 166, 181, 175, 171, 153, 158, 175, 173 and 157 cm⁻¹, respectively, which are similar to the typical radial breathing mode of B₄₀ at 170 cm⁻¹.⁶⁶ The breathing modes are used to identify the hollow structures in nanotubes. These typical radial breathing modes are close to the location of B₄₀, it suggests that the addition of metal atoms just lead to small change of B₄₀ configuration. However, Au@B₄₀⁻ has not the radial breathing mode near the frequency of 170 cm⁻¹ which further indicates that Au@B₄₀⁻ has obvious changes. The calculated results indicate that all the vibrational modes of exohedral MB₄₀⁰⁻ (M = Cu, Ag, and Au) and endohedral Cu@B₄₀⁰⁻ and Au@B₄₀⁻ are infrared active and Raman active. However, like the borospherene B₄₀,^{39,40} there are some silent modes in endohedral Ag@B₄₀⁰⁻ and Au@B₄₀, which means that these vibrational modes are Raman inactive and infrared inactive.



Table 1 The symmetries, relative energies (neutral CuB_{40} : the energy of CuB_{40} is set to be zero, anionic CuB_{40}^- : the energy of CuB_{40}^- is set to be zero, neutral AgB_{40} : the energy of AgB_{40} is set to be zero, anionic AgB_{40}^- : the energy of AgB_{40}^- is set to be zero, neutral AuB_{40} : the energy of AuB_{40} is set to be zero, anionic AuB_{40}^- : the energy of AuB_{40}^- is set to be zero), energy gaps (E_g), dipole moments (μ), lowest frequencies, and states of metalloborospherenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$). The superscripts a and b denote the alpha electron and beta electron, respectively

	Symmetry	Relative energy/eV	E_g /eV	μ /Debye	State	Lowest frequency/ cm^{-1}
CuB_{40}	C_s	0.00	1.40 ^a 3.00 ^b	0.82	$^2A'$	116
CuB_{40}^-	C_s	0.00	1.21	0.68	$^1A'$	107
$\text{Cu}@B_{40}$	C_s	1.04	1.53 ^a 2.88 ^b	0.39	$^2A'$	89
$\text{Cu}@B_{40}^-$	C_s	0.89	1.52	0.56	$^1A'$	92
AgB_{40}	C_s	0.00	2.12 ^a 2.52 ^b	3.78	$^2A'$	25
AgB_{40}^-	C_s	0.00	1.83	3.49	$^1A'$	36
$\text{Ag}@B_{40}$	D_{2d}	1.19	1.44 ^a 2.99 ^b	0.00	2B_2	47
$\text{Ag}@B_{40}^-$	D_{2d}	1.53	1.51	0.00	1A_1	52
AuB_{40}	C_s	0.00	2.45 ^a 2.21 ^b	0.65	$^2A'$	34
AuB_{40}^-	C_s	0.00	2.20	4.76	$^1A'$	44
$\text{Au}@B_{40}$	D_{2d}	2.50	1.80 ^a 3.14 ^b	0.00	2A_1	25
$\text{Au}@B_{40}^-$	C_1	2.90	2.30	2.46	1A	67

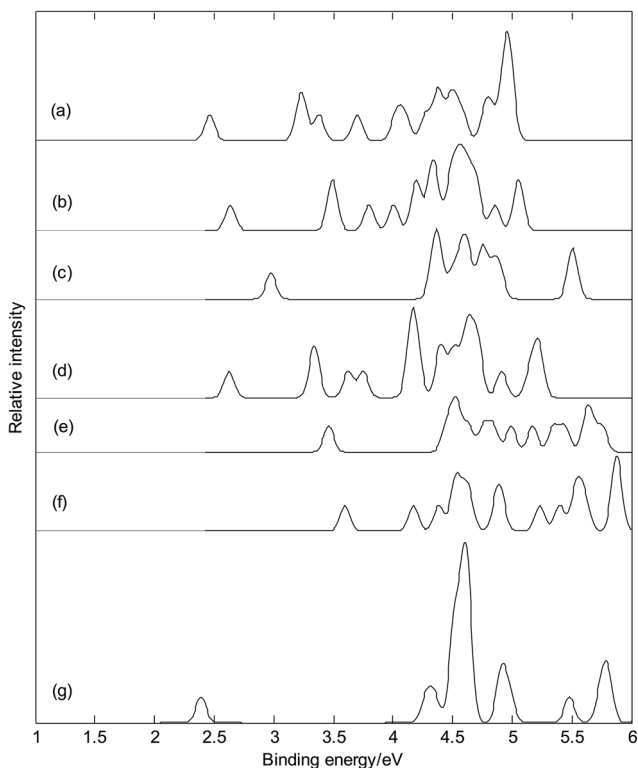


Fig. 11 Simulated photoelectron spectra of metalloborospherenes MB_{40}^- ($M = \text{Cu}, \text{Ag}, \text{and Au}$) and borospherene B_{40}^- . (a): $C_s \text{Cu}\text{B}_{40}^-$, (b): $C_s \text{Cu}@B_{40}^-$, (c): $C_s \text{Ag}\text{B}_{40}^-$, (d): $D_{2d} \text{Ag}@B_{40}^-$, (e): $C_s \text{Au}\text{B}_{40}^-$, (f): $C_1 \text{Au}@B_{40}^-$, (g): $D_{2d} \text{B}_{40}^-$. The simulations were done by fitting the distributions of calculated vertical detachment energies at the PBE0 level with unit-area Gaussian functions of 0.05 eV half-width.

Fig. 13(g and h) show that endohedral $\text{Ag}@B_{40}$ and $\text{Ag}@B_{40}^-$ have the similar Raman spectra, however, the addition of an electron weakens the main characteristic peaks. It's worth noting that the Raman spectra of endohedral $\text{Ag}@B_{40}^{0/-}$ are some

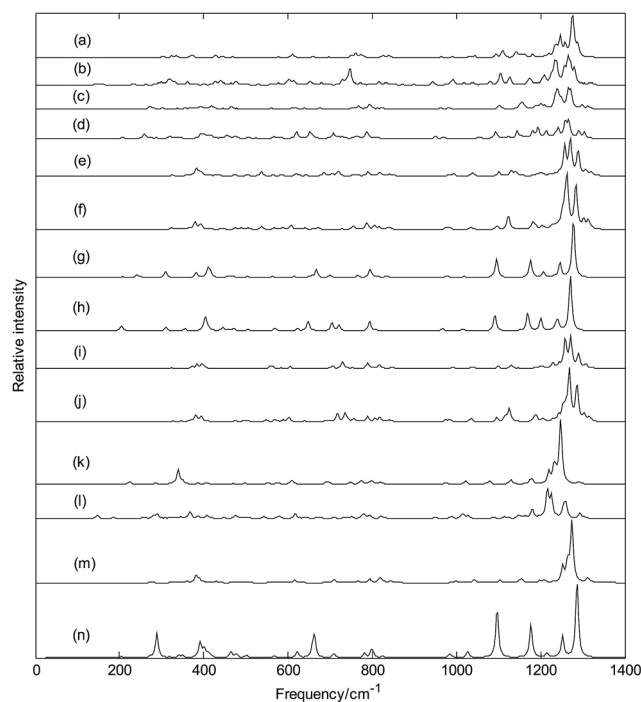


Fig. 12 Predicted infrared spectra of metalloborospherenes $\text{MB}_{40}^{0/-}$ ($M = \text{Cu}, \text{Ag}, \text{and Au}$) and borospherenes $\text{B}_{40}^{0/-}$. (a): $C_s \text{Cu}\text{B}_{40}$, (b): $C_s \text{Cu}\text{B}_{40}^-$, (c): $C_s \text{Cu}@B_{40}$, (d): $C_s \text{Cu}@B_{40}^-$, (e): $C_s \text{Ag}\text{B}_{40}$, (f): $C_s \text{Ag}\text{B}_{40}^-$, (g): $D_{2d} \text{Ag}@B_{40}$, (h): $D_{2d} \text{Ag}@B_{40}^-$, (i): $C_s \text{Au}\text{B}_{40}$, (j): $C_s \text{Au}\text{B}_{40}^-$, (k): $D_{2d} \text{Au}@B_{40}$, (l): $C_1 \text{Au}@B_{40}^-$, (m): $D_{2d} \text{B}_{40}$, (n): $D_{2d} \text{B}_{40}^-$.



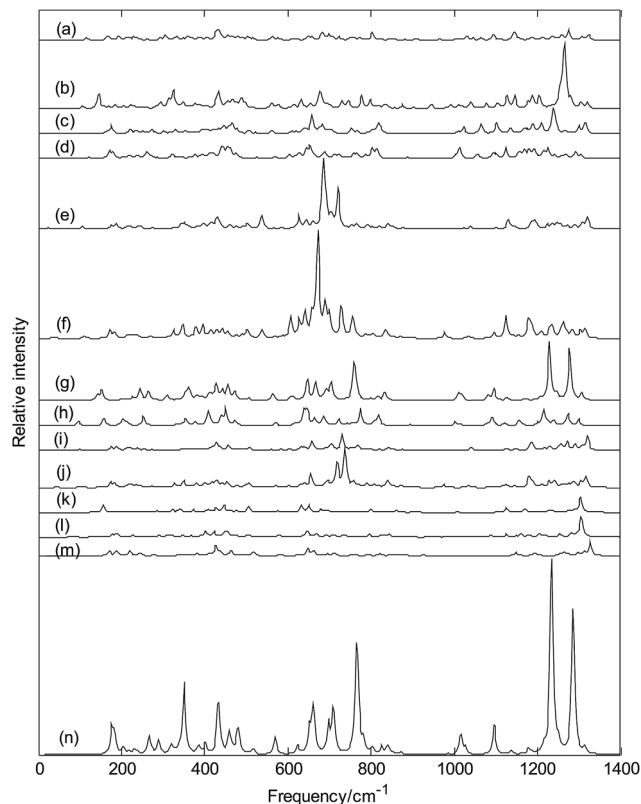


Fig. 13 Predicted Raman spectra of metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$) and borospherenes $B_{40}^{0/-}$. (a): $C_s Cu\text{@}B_{40}$, (b): $C_s Cu\text{@}B_{40}^-$, (c): $C_s Cu\text{@}B_{40}$, (d): $C_s Cu\text{@}B_{40}^-$, (e): $C_s Ag\text{@}B_{40}$, (f): $C_s Ag\text{@}B_{40}^-$, (g): $D_{2d} Ag\text{@}B_{40}$, (h): $D_{2d} Ag\text{@}B_{40}^-$, (i): $C_s Au\text{@}B_{40}$, (j): $C_s Au\text{@}B_{40}^-$, (k): $D_{2d} Au\text{@}B_{40}$, (l): $C_1 Au\text{@}B_{40}^-$, (m): $D_{2d} B_{40}$, (n): $D_{2d} B_{40}^-$.

similar to that of dianion B_{40}^{2-} (see Fig. 13(n)) and Raman spectra of endohedral $Au\text{@}B_{40}$ are some similar to that of B_{40} (see Fig. 13(m)).³⁴ Except for $Ag\text{@}B_{40}^{0/-}$ and $Au\text{@}B_{40}$, Fig. 13 indicates that doping of metal atom in B_{40} also changes the Raman peaks of B_{40} such as some weakened vibrational modes and some enhanced characteristic peaks.⁴⁰ In addition, Raman spectra of endohedral metalloborospherenes are different from that of corresponding exohedral metalloborospherenes, these spectral characteristics can be used to identify the endohedral and exohedral configurations.

Raman spectra, as the supplement of infrared spectra, can be used for the basis of identification of metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$). From the infrared and Raman spectra of each metalloborospherene, we can find, at some frequencies, infrared absorption peaks are strong, but the Raman peaks are very weak. However, at some frequencies, infrared absorption peaks are very weak (or infrared inactive), but the Raman peaks are strong. In addition, at some frequencies, both the infrared and Raman peaks are strong. Infrared intensity is proportional to $|\partial\bar{\mu}/\partial Q_i|^2$,^{67,68} where $\bar{\mu}$ is the electric dipole moment and Q_i is the normal-mode coordinate. A vibrational mode of molecular with no change of dipole moment is infrared inactive, we can't obtain the normal mode frequency from the infrared spectral data in experiments.

However, this vibrational mode may lead to the change of polarizability, this indicates that the vibrational mode is Raman active. The calculated Raman spectra can be useful for analytical purposes and contribute significantly to spectral interpretation and vibrational assignments, also can provide technical guidance for future synthesis.

Electronic absorption spectra

Finally, we calculated electronic absorption spectra of exohedral metalloborospherenes $M\text{&}B_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$), as shown in Fig. 14. The previous research results show that B_{40} has only UV-Vis absorption bands (see Fig. 14(g)),^{39,40} however, one can observe several near infrared (NIR) absorption bands of metalloborospherenes $M\text{&}B_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$) from 780 to 3000 nm. It suggests that doping of metal atom in B_{40} can redshift the electronic spectra. Fig. 14(a) presents the electronic spectra of $Cu\text{&}B_{40}$, the largest excitation wavelength (first excited state) is 2175 nm. The absorption band in region NIR from 1500 to 2500 nm comes from the first three excited states which originate from the electron transitions from α -HOMO to α -LUMO/ α -LUMO+1/ α -LUMO+2. Fig. 14(b) presents the electronic absorption spectra of $Cu\text{&}B_{40}^-$, the largest excitation wavelength is 2701 nm. The absorption band in region NIR from 2500 to 3000 nm comes from the first excited state which

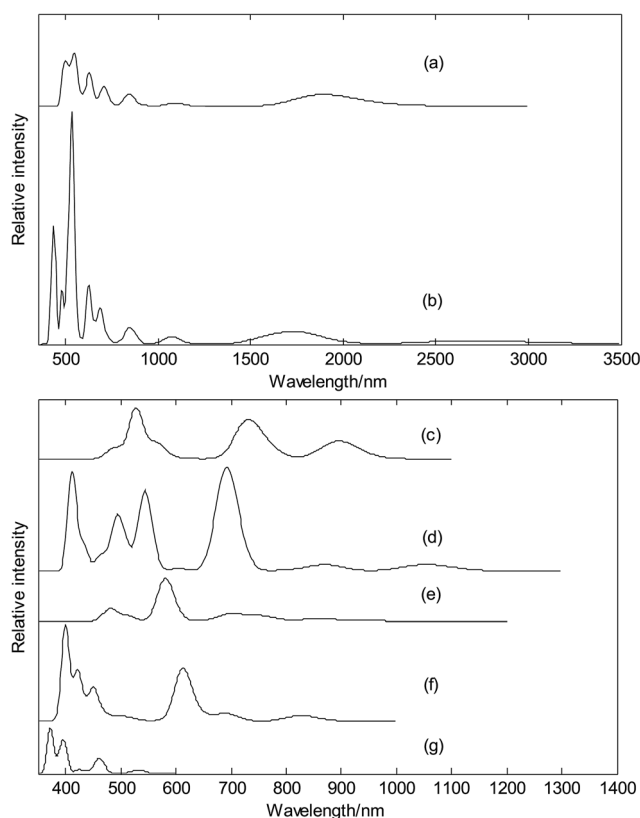


Fig. 14 Predicted electronic absorption spectra of exohedral metalloborospherenes $MB_{40}^{0/-}$ ($M = Cu, Ag, \text{ and } Au$) and borospherene B_{40} . (a): $C_s Cu\text{@}B_{40}$, (b): $C_s Cu\text{@}B_{40}^-$, (c): $C_s Ag\text{@}B_{40}$, (d): $C_s Ag\text{@}B_{40}^-$, (e): $C_s Au\text{@}B_{40}$, (f): $C_s Au\text{@}B_{40}^-$, (g): $D_{2d} B_{40}$.



originates from the electron transitions from HOMO to LUMO. Fig. 14(a and b) indicate that electronic spectra of $\text{Cu}\&\text{B}_{40}^-$ are apparently red-shifted comparing with $\text{Cu}\&\text{B}_{40}$. Fig. 14(c) presents the electronic spectra of the $\text{Ag}\&\text{B}_{40}$. The largest excitation wavelength is 898 nm. The absorption band in region NIR from 800 to 1000 nm comes from the first two excited states which originate from the electron transitions of $\alpha\text{-HOMO} \rightarrow \alpha\text{-LUMO}/\alpha\text{-LUMO}+1$ coupled with $\beta\text{-HOMO}/\beta\text{-HOMO}-1/\beta\text{-HOMO}-2/\beta\text{-HOMO}-7 \rightarrow \beta\text{-LUMO}$. Fig. 14(d) presents the electronic spectra of the $\text{Ag}\&\text{B}_{40}^-$. The largest excitation wavelength is 1054 nm. The absorption band in region NIR from 1000 to 1200 nm comes from the first excited state which originates from the electron transitions from HOMO to LUMO/LUMO+2. Fig. 14(e) presents the electronic absorption spectra of the $\text{Au}\&\text{B}_{40}$, the largest excitation wavelength is 988 nm. The weak absorption band in region NIR from 800 to 1000 nm comes from the first four excited states which originate from the electron transitions of $\alpha\text{-HOMO} \rightarrow \alpha\text{-LUMO}+1$ coupled with $\beta\text{-HOMO}/\beta\text{-HOMO}-1/\beta\text{-HOMO}-2/\beta\text{-HOMO}-3/\beta\text{-HOMO}-4 \rightarrow \beta\text{-LUMO}$. Fig. 14(f) presents the electronic absorption spectra of the $\text{Au}\&\text{B}_{40}^-$, the largest excitation wavelength is 828 nm. The absorption band in region NIR from 780 to 900 nm comes from the first excited state which originates from the electron transitions from HOMO to LUMO. Other NIR absorption bands of metalloborosphenes $\text{MB}_{40}^{0/-}$ ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) come from the higher excited states which originate from the electron transitions of corresponding orbitals.

For the metalloborosphenes $\text{M}\&\text{B}_{40}^-$ ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) with closed-shell electronic structure, the minimum excitation energy (the largest excitation wavelength) mainly comes from the electron transition from HOMO to LUMO. HOMO–LUMO energy gap reflects the probability of the molecules jumping from ground state to excited state. Generally speaking, the larger energy gap can lead to the larger electron excitation energy, *i.e.*, the smaller the probability of electronic transition. On the contrary, the molecule with smaller energy gap is easier to jump to the excited state. According to the previous results, the HOMO–LUMO energy gaps are 1.2113, 1.8272 and 2.1975 eV for $\text{Cu}\&\text{B}_{40}^-$, $\text{Ag}\&\text{B}_{40}^-$, and $\text{Au}\&\text{B}_{40}^-$, respectively. Although the energy gap of ground state does not represent the minimum excitation energy, the increasing HOMO–LUMO energy gaps just reflect the decreasing largest excitation wavelengths 2701, 1054 and 828 nm for $\text{Cu}\&\text{B}_{40}^-$, $\text{Ag}\&\text{B}_{40}^-$, and $\text{Au}\&\text{B}_{40}^-$, respectively. The calculated electronic spectra may be used for the structural analysis in conjunction with other techniques.

Conclusions

In a summary, the structures, stabilities, photoelectron spectra, infrared spectra, Raman spectra, and electronic absorption spectra of metalloborosphenes MB_{40}^- ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) have been studied at the level of density functional theory (DFT) and time-dependent density functional theory (TD-DFT). Potential energy scan confirms that there are two possible ground state geometries where dopant can exist outside or inside cage. In addition, potential energy scan

confirms metalloborosphenes MB_{40}^- ($\text{M} = \text{Ag}$ and Au) have a very high energy barrier around the surface of B_{40} and the energy barrier can confine dopant inside B_{40} . The further calculated results suggest that $\text{Cu}, \text{Ag}, \text{and Au}$ atoms can form stable exohedral metalloborosphenes $\text{M}\&\text{B}_{40}^{0/-}$ ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) and endohedral metalloborosphenes $\text{M}\&\text{B}_{40}^{0/-}$ ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$). In addition, relative energies of these metalloborosphenes reveal that the $\text{Cu}, \text{Ag}, \text{and Au}$ atoms favor the exohedral configuration. The calculated photoelectron spectra show that metalloborosphenes MB_{40}^- ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) have different and meaningful photoelectron spectra which can be used as theoretical basis for the identification of metalloborosphenes MB_{40}^- ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$). The calculated infrared spectra suggest that location of dopant and charge of metalloborosphenes play an important role in infrared spectra. In addition, the calculated electronic spectra indicate that metalloborosphenes MB_{40}^- ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) have obvious near-IR absorption peaks. These spectral features are readily compared with future spectroscopy measurements and can be used as fingerprints to identify and distinguish the metalloborosphenes MB_{40}^- ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$). The metalloborosphenes $\text{MB}_{40}^{0/-}$ ($\text{M} = \text{Cu}, \text{Ag}, \text{and Au}$) have provided an important clue for the development of new boron-based materials such as molecular device.

Conflict of interest

There are no conflicts of interest to declare.

Acknowledgements

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Notes and references

- 1 H. W. Kroto, J. R. Heath, S. C. O'Brien, R. F. Curl and R. E. Smalley, *Nature*, 1985, **318**, 162.
- 2 H. J. Zhai, B. Kiran, J. Li and L. S. Wang, *Nat. Mater.*, 2003, **2**, 827.
- 3 I. A. Popov, Z. A. Piazza, W. L. Li, L. S. Wang and A. I. Boldyrev, *J. Chem. Phys.*, 2013, **139**, 144307.
- 4 Z. A. Piazza, H. S. Hu, W. L. Li, Y. F. Zhao, J. Li and L. S. Wang, *Nat. Commun.*, 2014, **5**, 3113.
- 5 A. P. Sergeeva, D. Y. Zubarev, H. J. Zhai, A. I. Boldyrev and L. S. Wang, *J. Am. Chem. Soc.*, 2008, **130**, 7244.
- 6 W. L. Li, Q. Chen, W. J. Tian, H. Bai, Y. F. Zhao, H. S. Hu, J. Li, H. J. Zhai, S. D. Li and L. S. Wang, *J. Am. Chem. Soc.*, 2014, **136**, 12257.
- 7 E. Oger, N. R. Crawford, R. Kelting, P. Weis, M. M. Kappes and R. Ahlrichs, *Angew. Chem., Int. Ed.*, 2007, **46**, 8503.
- 8 A. N. Alexandrova, A. I. Boldyrev, H. J. Zhai and L. S. Wang, *Coord. Chem. Rev.*, 2006, **250**, 2811.



- 9 A. P. Sergeeva, I. A. Popov, Z. A. Piazza, W. L. Li, C. Romanescu, L. S. Wang and A. I. Boldyrev, *Acc. Chem. Res.*, 2014, **47**, 1349.
- 10 N. G. Szwacki, A. Sadrzadeh and B. I. Yakobson, *Phys. Rev. Lett.*, 2007, **98**, 166804.
- 11 X. L. Sheng, Q. B. Yan, Q. R. Zheng and G. Su, *Phys. Chem. Chem. Phys.*, 2009, **11**, 9696.
- 12 L. Wang, J. J. Zhao, F. Y. Li and Z. F. Chen, *Chem. Phys. Lett.*, 2010, **501**, 16.
- 13 C. Özdoğan, S. Mukhopadhyay, W. Hayami, Z. B. Güvenç, R. Pandey and I. Boustani, *J. Phys. Chem. C*, 2010, **114**, 4362.
- 14 L. J. Cheng, *J. Chem. Phys.*, 2012, **136**, 104301.
- 15 H.-G. Lu and S. D. Li, *J. Chem. Phys.*, 2013, **139**, 224307.
- 16 J. Lv, Y. C. Wang, L. Zhu and Y. M. Ma, *Nanoscale*, 2014, **6**, 11692–11696.
- 17 H. T. Pham, L. V. Duong and M. T. Nguyen, *J. Phys. Chem. C*, 2014, **118**, 24181.
- 18 L. V. Duong, H. T. Pham, N. M. Tam and M. T. Nguyen, *Phys. Chem. Chem. Phys.*, 2014, **16**, 19470.
- 19 T. B. Tai, L. V. Duong, H. T. Pham, D. T. T. Maia and M. T. Nguyen, *Chem. Commun.*, 2014, **50**, 1558.
- 20 H. Tang and S. Ismail-Beigi, *Phys. Rev. Lett.*, 2007, **99**, 115501.
- 21 D. L. V. K. Prasad and E. D. Jemmis, *Phys. Rev. Lett.*, 2008, **100**, 165504.
- 22 D. L. V. K. Prasad and E. D. Jemmis, *Appl. Phys. Lett.*, 2010, **96**, 023108.
- 23 H. J. Zhai, Y. F. Zhao, W. L. Li, Q. Chen, H. Bai, H. S. Hu, Z. A. Piazza, W. J. Tian, H. G. Lu, Y. B. Wu, Y. W. Mu, G. F. Wei, Z. P. Liu, J. Li, S. D. Li and L. S. Wang, *Nat. Chem.*, 2014, **6**, 727.
- 24 Q. Chen, W. L. Li, Y. F. Zhao, S. Y. Zhang, H. S. Hu, H. Bai, H. R. Li, W. J. Tian, H. G. Lu, H. J. Zhai, S. D. Li, J. Li and L. S. Wang, *ACS Nano*, 2015, **9**, 754.
- 25 G. Martínez-Guajardo, J. L. Cabellos, A. Díaz-Celaya, S. Pan, R. Islas, P. K. Chattaraj, T. Heine and G. Merino, *Sci. Rep.*, 2015, **5**, 11287.
- 26 H. L. Dong, T. J. Hou, S. T. Lee and Y. Y. Li, *Sci. Rep.*, 2015, **5**, 09952.
- 27 I. A. Popov, T. Jian, G. V. Lopez, A. I. Boldyrev and L. S. Wang, *Nat. Commun.*, 2015, **6**, 8654.
- 28 T. Jian, W. L. Li, I. A. Popov, G. V. Lopez, X. Chen, A. I. Boldyrev, J. Li and L. S. Wang, *J. Chem. Phys.*, 2016, **144**, 154310.
- 29 Y. J. Wang, Y. F. Zhao, W. L. Li, T. Jian, Q. Chen, X. R. You, T. Ou, X. Y. Zhao, H. J. Zhai, S. D. Li, J. Li and L. S. Wang, *J. Chem. Phys.*, 2016, **144**, 064307.
- 30 H. R. Li, T. Jian, C. Q. Miao, Y. J. Wang, Q. Chen, X. M. Luo, K. Wang, H. J. Zhai, S. D. Li and L. S. Wang, *Phys. Chem. Chem. Phys.*, 2016, **18**, 29147.
- 31 W. L. Li, R. Pal, Z. A. Piazza, X. C. Zeng and L. S. Wang, *J. Chem. Phys.*, 2015, **142**, 204305.
- 32 T. B. Tai and M. T. Nguyen, *Chem. Commun.*, 2016, **52**, 1653.
- 33 S. X. Li, Z. P. Zhang, Z. W. Long and S. J. Qin, *Sci. Rep.*, 2017, **7**, 40081.
- 34 T. B. Tai and M. T. Nguyen, *RSC Adv.*, 2017, **7**, 22243.
- 35 T. B. Tai and M. T. Nguyen, *Chem. Commun.*, 2015, **51**, 7677.
- 36 S. G. Xu, Y. J. Zhao, J. H. Liao and X. B. Yang, *J. Chem. Phys.*, 2015, **142**, 214307.
- 37 N. Karmodak and E. D. Jemmis, *Chem.–Asian J.*, 2016, **11**, 3350.
- 38 H. L. Dong, B. Lin, K. Gilmore, T. J. Hou, S. T. Lee and Y. Y. Li, *Curr. Appl. Phys.*, 2015, **15**, 1084.
- 39 R. X. He and X. C. Zeng, *Chem. Commun.*, 2015, **51**, 3185.
- 40 S. X. Li, Z. P. Zhang, Z. W. Long, G. Y. Sun and S. J. Qin, *Sci. Rep.*, 2016, **6**, 25020.
- 41 S. Bosi, R. T. Da, G. Spalluto and M. Prato, *Eur. J. Med. Chem.*, 2003, **38**, 913.
- 42 R. Bakry, R. M. Vallant, M. Najam-ul-Haq, M. Rainer, Z. Szabo, C. W. Huck and G. K. Bonn, *Int. J. Nanomed.*, 2007, **2**, 639.
- 43 P. W. M. Blom, V. D. Mihailetchi, L. J. A. Koster and D. E. Markov, *Adv. Mater.*, 2007, **19**, 1551.
- 44 B. Liu, R. Q. Png, L. H. Zhao, L. L. Chua, R. H. Friend and P. K. Ho, *Nat. Commun.*, 2012, **3**, 1321.
- 45 V. Buntar and H. W. Weber, *Supercond. Sci. Technol.*, 1996, **9**, 599.
- 46 Y. Chai, T. Guo, C. M. Jin, R. E. Haufler, L. P. F. Chibante, J. Fure, L. H. Wang, J. M. Alford and R. E. Smalley, *J. Phys. Chem.*, 1991, **95**, 7564.
- 47 D. S. Bethune, R. D. Johnson, J. R. Salem, M. S. Vries and C. S. Yannoni, *Nature*, 1993, **366**, 123.
- 48 T. Hirata, R. Hatakeyama, T. Mieno and N. Sato, *J. Vac. Sci. Technol.*, A, 1996, **14**, 615.
- 49 M. N. Chaur, F. Melin, A. L. Ortiz and L. Echegoyen, *Angew. Chem., Int. Ed.*, 2009, **48**, 7514.
- 50 H. Bai, Q. Chen, H. J. Zhai and S. D. Li, *Angew. Chem., Int. Ed.*, 2015, **54**, 941.
- 51 P. Jin, Q. H. Hou, C. C. Tang and Z. F. Chen, *Theor. Chem. Acc.*, 2015, **134**, 13.
- 52 W. Fa, S. Chen, S. Pande and X. C. Zeng, *J. Phys. Chem. A*, 2015, **119**, 11208.
- 53 Q. Chen, T. Gao, W. J. Tian, H. Bai, S. Y. Zhang, H. R. Li, C. Q. Miao, Y. W. Mu, H. G. Lu, H. J. Zhai and S. D. Li, *Phys. Chem. Chem. Phys.*, 2015, **17**, 19690.
- 54 Q. Chen, H. R. Li, C. Q. Miao, Y. J. Wang, H. G. Lu, Y. W. Mu, G. M. Ren, H. J. Zhai and S. D. Li, *Phys. Chem. Chem. Phys.*, 2016, **18**, 11610.
- 55 Q. Chen, H. R. Li, W. J. Tian, H. G. Lu, H. J. Zhai and S. D. Li, *Phys. Chem. Chem. Phys.*, 2016, **18**, 14186.
- 56 W. J. Tian, Q. Chen, H. R. Li, M. Yan, Y. W. Mu, H. G. Lu, H. J. Zhai and S. D. Li, *Phys. Chem. Chem. Phys.*, 2016, **18**, 9922.
- 57 P. Jin, L. Yang, C. Liu, Q. H. Hou and L. L. Li, *Theor. Chem. Acc.*, 2017, **136**, 56.
- 58 W. Wang, Y. D. Guo and X. H. Yan, *RSC Adv.*, 2016, **6**, 40155.
- 59 Y. P. An, M. J. Zhang, D. P. Wu, Z. M. Fu, T. T. Wang and C. X. Xia, *Phys. Chem. Chem. Phys.*, 2016, **18**, 12024.
- 60 Z. Yang, Y. L. Ji, G. Q. Lan, L. C. Xu, X. G. Liu and B. S. Xu, *Solid State Commun.*, 2015, **217**, 38.
- 61 M. Moradi, Z. Bagheri and A. Bodaghi, *Phys. E*, 2017, **89**, 148.
- 62 M. Moradi, V. Vahabi and A. Bodaghi, *J. Mol. Liq.*, 2016, **223**, 315.



- 63 Y. Noguchi, O. Sugino, H. Okada and Y. Matsuo, *J. Phys. Chem. C*, 2013, **117**, 15362.
- 64 R. Bauernschmitt and R. Ahlrichs, *Chem. Phys. Lett.*, 1996, **256**, 454.
- 65 M. J. Frisch, *et al.*, *Gaussian 09, Revision A.02*, Gaussian Inc., Wallingford, CT, 2009.
- 66 Q. Chen, S. Y. Zhang, H. Bai, W. J. Tian, T. Gao, H. R. Li, C. Q. Miao, Y. W. Mu, H. G. Lu, H. J. Zhai and S. D. Li, *Angew. Chem., Int. Ed.*, 2015, **54**, 8160.
- 67 Y. Bai, X. Y. Zhao, T. W. Li, Z. L. Lv, S. J. Lv, H. Han, Y. F. Yin and H. Wang, *Carbon*, 2014, **78**, 70.
- 68 D. Porezag and M. R. Pederson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 7830.

