



Cite this: *RSC Adv.*, 2017, 7, 43403

Highly uniform hierarchical Zn₂SnO₄ microspheres for the construction of high performance dye-sensitized solar cells†

Xin Wang,^{ab} Yu-Fen Wang,^{*ab} Qiu-Ping Luo,^c Jian-Hua Ren,^{ab} De-Jun Li^{ab} and Xi-Fei Li^{ID}^{*ab}

Received 21st June 2017
Accepted 30th August 2017

DOI: 10.1039/c7ra06906e

rsc.li/rsc-advances

3D hierarchical Zn₂SnO₄ microspheres show great potential as alternative photoanodes for dye-sensitized solar cells (DSSCs). Herein, hierarchical Zn₂SnO₄ microspheres with different sizes are fabricated *via* a one-step hydrothermal method, and a plausible formation process for Zn₂SnO₄ microspheres is proposed. The DSSCs devices based on 1.20 μm Zn₂SnO₄ microspheres show the best power conversion efficiency (PCE) of 4.00%. Further, TiCl₄ treatment of a Zn₂SnO₄ microsphere photoanode results in an improved PCE up to 4.72%.

1. Introduction

Since the pioneering report of dye-sensitized solar cells (DSSCs) by M. Grätzel in 1991, tremendous progress has been witnessed in this field over the past decades.^{1,2} So far, the highest power conversion efficiencies (PCEs) of more than 14% have been demonstrated.³ The photoanodes, composed of wide-band-gap semiconducting metal oxide films (such as TiO₂, ZnO, SnO₂ and Zn₂SnO₄) with a large internal surface area, play a key role in affecting the photovoltaic performance of DSSCs.^{4–11} Ternary oxide semiconductors, such as Zn₂SnO₄, have attracted worldwide attention due to their promising optical and electrical properties. Firstly, Zn₂SnO₄ possesses a wide band gap of 3.8 eV (TiO₂: 3.2 eV), whose conduction band position is similar to that of TiO₂ and ZnO. However, for these two metal oxides, TiO₂ and ZnO, the UV region of the solar spectrum suffers from photo-bleaching when applied as photoanodes in DSSCs;¹² secondly, Zn₂SnO₄ with a higher electron mobility of 10–15 cm² V⁻¹ s⁻¹ than TiO₂ (10–5 cm² V⁻¹ s⁻¹) implies more efficient electron transport characteristic in DSSCs.^{13,14} Thirdly, unlike ZnO, for which the Zn-dye + complex can decrease the photocurrent (*J*_{sc}) and photovoltage (*V*_{oc}), Zn₂SnO₄ shows better chemical stability in acid/basic solution.¹⁵ Last but not least, Zn₂SnO₄ possesses tunable work function and electric resistivity. It is well-known that the hierarchical structured materials have several

advantages while are applied in DSSCs, for instance, the large specific surface area could induce efficient dye adsorption, outstanding light-scattering ability could induce efficient light harvesting, and well-interconnected structures could induce boosted electron transport and suppressed electron recombination.^{16–18} But, fabrication of Zn₂SnO₄ microspheres with uniform morphological features pose the greatest challenge in their preparation because such ternary oxide needs high temperature (>200 °C) to crystallize and regulate the Zn and Sn ions during synthetic process.

In this work, we demonstrate a facile one-step hydrothermal process to fabricate 3D hierarchically Zn₂SnO₄ microspheres consisting of nanoplates, which could offer not only larger surface areas but also would result in a better light-scattering ability as well as electricity-generation properties. In addition, TiO₂-coated hierarchical Zn₂SnO₄ microspheres (~1.2 μm in diameter) are used as photoanode materials for improved PCE in DSSCs. The highest PCE in this work up to 4.72% has been achieved.

2. Experimental

2.1 Synthesis

Hierarchical Zn₂SnO₄ microspheres are synthesized *via* a one-step hydrothermal process. Typically, 1.402 g SnCl₄·5H₂O (A.R.), 1.756 g Zn(CH₃COO)₂·2H₂O (A.R.), 1.776 g NH₄F (A.R.), 1.480 g H₃BO₃ power (A.R.) are dissolved in 80 mL distilled H₂O in a 100 mL Teflon-lined container. Then, 2.0 g NaOH is added into the above solution with vigorous stirring. After continuously stirring for another 30 min, the obtained precursor solution is transferred to a stainless steel autoclave. After being sealed, the autoclave is kept heated at 200 °C for 24 h. While being cooled to the room temperature, the obtained solution is purified by centrifuging using 6000 rpm for 5 min to get the

^aEnergy & Materials Engineering Center, College of Physics and Materials Science, Tianjin Normal University, Tianjin 300387, China. E-mail: xfli@mail.tjnu.edu.cn; yfwang@mail.tjnu.edu.cn

^bTianjin International Joint Research Center of Surface Technology for Energy Storage Materials, Tianjin 300387, China

^cLaboratory of Optical Information Technology, School of Science, Wuhan Institute of Technology, Wuhan 430205, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra06906e



precipitates. During the centrifuge process, the precipitates are washed 3 times with absolute ethanol and distilled water in an ultrasonic cleaning bath for 10 min, respectively. The final white-color power is dried at 70 °C for the further characterizations. In order to tune the diameter of the hierarchical Zn_2SnO_4 microspheres, diethylene glycol (DEG) and ethylene glycol (EG) are used as the mixed solvent to support the Zn_2SnO_4 crystal growth.

2.2 Preparation of the Zn_2SnO_4 microspheres photoanode

First, the FTO glass (sheet resistance: 15 Ω /square, Nippon Sheet Glass, Japan) is ultrasonically cleaned with water, HCl (A.R.), acetone (A.R.) and ethanol (A.R.) for 30 min, successively. 1.0 g hierarchical Zn_2SnO_4 samples are ground for 30 min in the mixtures (8.0 mL ethanol, 3.0 g terpeneol, 0.2 mL acetic acid and 0.5 g ethyl cellulose) to form slurry, and then the obtained mixtures are sonicated in an ultrasonic bath for 15 min, finally to form Zn_2SnO_4 paste. The preparation of Zn_2SnO_4 paste is the same to the previous report. The obtained different Zn_2SnO_4 paste is coated on the cleaned FTO glass *via* the screen-printing technique. Meanwhile, the thickness of Zn_2SnO_4 film is adjusted through repeating the screen-printing times. Then, the obtained different Zn_2SnO_4 films are heated at 500 °C for 1.0 h in air. After that, the obtained different Zn_2SnO_4 films are immersed into a 40 mM TiCl_4 aqueous solution for another 30 min at 70 °C, followed by annealing at 520 °C for 30 min. When cooling down to room temperature, the fabricated Zn_2SnO_4 photoanodes are soaked into N719 dye solution ($\text{Ru}[\text{LL}'-(\text{NCS})_2]$, $\text{L} = 2,2'$ -bipyridyl-4,4'-dicarboxylic acid, $\text{L} = 2,2'$ -bipyridyl-4,4'-ditetrabutylammonium carboxylate, 5.0×10^{-4} M, Solarconix Co.) in acetonitrile/*tert*-butanol (1 : 1, v/v) for 24 h. The Pt counter electrode is prepared *via* adding H_2PtCl_6 solution (5.0×10^{-4} M) on the surface of the cleaned FTO glass, and then heat at 400 °C for 15 min in air. And the electrolyte is consisted of I_2 (0.03 M), 1-propyl-3-methyl-imidazolium iodide (PMII, 0.6 M), LiI (0.05 M), guanidine thiocyanate (GuNCS, 0.1 M, Aldrich), and 4-*tert*-butylpyridine (*t*-BP, 0.5 M, Aldrich) in a mixture contained acetonitrile and valeronitrile (85 : 15, v/v). And the active area of the N719 dye-sensitized Zn_2SnO_4 films in this work is about 0.152 cm^2 .

2.3 Characterization

The phase purity of the products is characterize by powder X-ray diffraction (XRD) (Bruker D8 Advance X-ray diffractometer, Cu $\text{K}\alpha$ radiation, $\lambda = 1.5418$ Å). The morphology is investigated by field emission scanning electron microscopy (FE-SEM, SU8010) and transmission electron microscope (TEM, JEOL-2010). And the thickness of Zn_2SnO_4 film is determined *via* a profilometer (Ambios, XP-2). UV-vis diffused reflectance spectra is investigated on a UV-vis-NIR Spectrophotometer (Shimadzu UV-3600). The current–voltage (J – V) characteristics in this work are performed with Keithley 2400 source meter, (AM 1.5 G illumination, 100 mW cm^{-2}), which are provided by solar simulator (69 920, 1 kW Xe lamp, Oriol). The standard Si solar cell with different optical filter (Oriol) provides various irradiance intensities (from 0.1 to 1.0 sun), is introduced to calibrate the

incident light intensity. Incident photon-to-current conversion efficiency (IPCE) signal is recorded on a Keithley 2000 multi-meter under the illumination of a 150 W tungsten lamp with a Spectral Product DK240 monochromator. The intensity-modulated photovoltage spectroscopy (IMVS) and intensity-modulated photocurrent spectroscopy (IMPS) measurements (two electrode polarization scheme) are carried on an electrochemical workstation (Zahner, Zennium), under the modulated blue light emitting diodes (457 nm) driven by a Zahner (PP211) source supply, the electrochemical workstation with a frequency response analyzer. The electrochemical impedance spectroscopy (EIS) is also performed on an electrochemical workstation (Zahner, Zennium) in dark (bias potential: -0.70 V, frequency range: 10 mHz to 1 MHz).

3. Results and discussion

In this study, $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ and $\text{SnCl}_4 \cdot 5\text{H}_2\text{O}$ are used as precursors in the hydrothermal process. The phase structures of prepared samples are characterized by XRD. As shown in Fig. 1a, all peaks are well consistent with pure cubic structure of Zn_2SnO_4 with the lattice constants of $a = b = c = 8.657$ Å (JCPDS No. 24-1470). And the strong signal of characteristic peaks indicates the high crystallinity of as-prepared Zn_2SnO_4 microspheres. As seen in Fig. 1b, the obtained Zn_2SnO_4 microspheres have a uniform size distribution with diameter of 1.20 μm . Fig. 1c shows the hierarchical structure of as-prepared Zn_2SnO_4 microsphere with a relatively rough surface. Specifically, the microsphere composes of numerous cross-linked and interconnected nanosheets. Fig. 1d confirms the microscale of the Zn_2SnO_4 spheres.

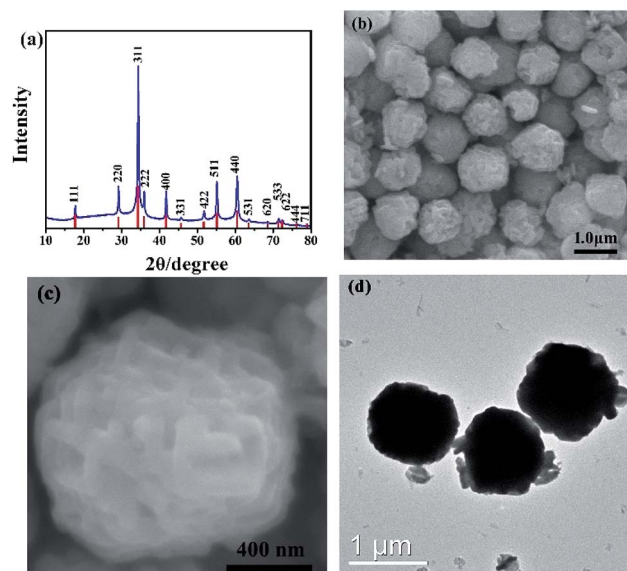


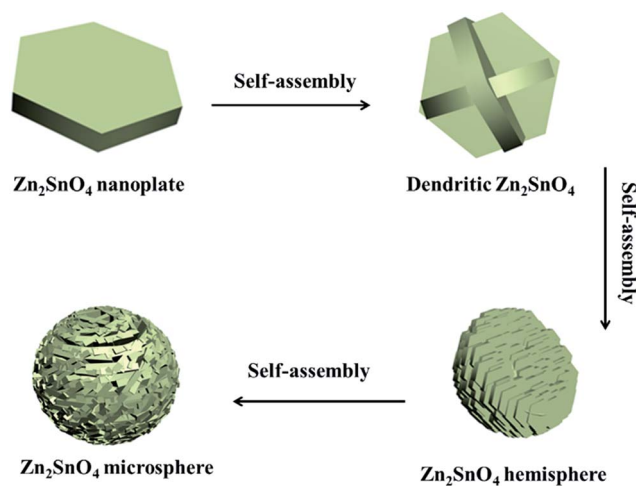
Fig. 1 (a) XRD patterns of as-synthesized Zn_2SnO_4 microspheres after hydrothermal reaction for 24 h. (b) Low-magnification and (c) high-magnification FE-SEM images of as-prepared Zn_2SnO_4 microsphere. (d) TEM image of Zn_2SnO_4 microspheres.



The time-dependent experiments are carried out to explore the detail formation process of the hierarchical Zn_2SnO_4 microspheres. As shown in Fig. 2, the morphologies of the Zn_2SnO_4 are delicately dependent on the hydrothermal reaction time. An interesting morphological evolution has been observed (Fig. 2, 1b and c) since reaction time increased from 1 h to 24 h. At the early stage (1 h), the Zn_2SnO_4 nanoplates are observed (Fig. 2a). Increasing the time to 2 h, the primary nanoplates occur to connect with each other to form a cross-shaped Zn_2SnO_4 structure (Fig. 2b). Prolonging the hydrothermal reaction time to 4 h, one can observe the formation of the irregular Zn_2SnO_4 hemispheres ($\sim 0.6 \mu\text{m}$ in diameter), which consist of numerous interconnected nanosheets (Fig. 2c). The semi-matured hierarchical Zn_2SnO_4 microspheres ($\sim 1.0 \mu\text{m}$ in diameter) can be obtained when the hydrothermal reaction increases to 12 h, as shown in Fig. 2d. Further increasing the hydrothermal time to 24 h leads to the formation of well-defined hierarchical Zn_2SnO_4 microspheres ($\sim 1.20 \mu\text{m}$ in diameter) which are made up of cross-linked and interconnected nanosheets (Fig. 1b and c).

Based on the above discussion, a possible growth process of the hierarchical Zn_2SnO_4 microspheres is proposed (Scheme 1). During the initial stage, the Zn and Sn precursor results in the self-assembly formation of primary Zn_2SnO_4 nanoplates, and then the cross-shaped Zn_2SnO_4 structure is formed *via* the interconnection between nanoplates. With further aggregation of these Zn_2SnO_4 nanoplates, an irregular Zn_2SnO_4 hemisphere is formed. Gradually, the newly formed nanoplates selectively attach onto the surface of existing secondary structure, which evolves to the final 3D hierarchical Zn_2SnO_4 microspheres *via* the self-assembly process.

The preliminary study proved the diethylene glycol (DEG) can influence the sizes of the samples.⁸ Hence, in order to tune the diameter of the hierarchical Zn_2SnO_4 microspheres, DEG and EG are used as the mixed solvent to support the Zn_2SnO_4



Scheme 1 Schematic illustration of the formation process of the as-prepared hierarchical Zn_2SnO_4 microspheres.

crystal growth. Meanwhile, it is found that the composition of solvent significantly affects the sizes of Zn_2SnO_4 microspheres. The FE-SEM and TEM images of the Zn_2SnO_4 spheres prepared under the different solvent conditions, as shown in Fig. 3. Typically, EG/ H_2O (20 : 60 in volume ratio) and DEG/ H_2O (20 : 60 in volume ratio) lead to the formation of hierarchical Zn_2SnO_4 microspheres with average size of $0.85 \mu\text{m}$ and $0.60 \mu\text{m}$, respectively, which are smaller than that of pure H_2O -based counterpart ($1.20 \mu\text{m}$). According to the previous investigation results, the growth rates of crystals vary along the different crystallographic directions, so that the final morphology is controlled by the fastest growth plane.¹⁹ The influences of EG

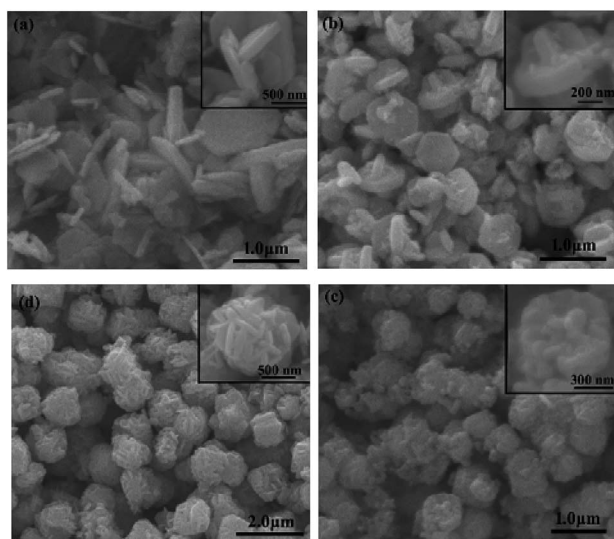


Fig. 2 FE-SEM images of the Zn_2SnO_4 microspheres synthesized at different hydrothermal time: (a) 1 h, (b) 2 h, (c) 4 h, (d) 12 h. The inset images are the corresponding high-magnification FE-SEM images.

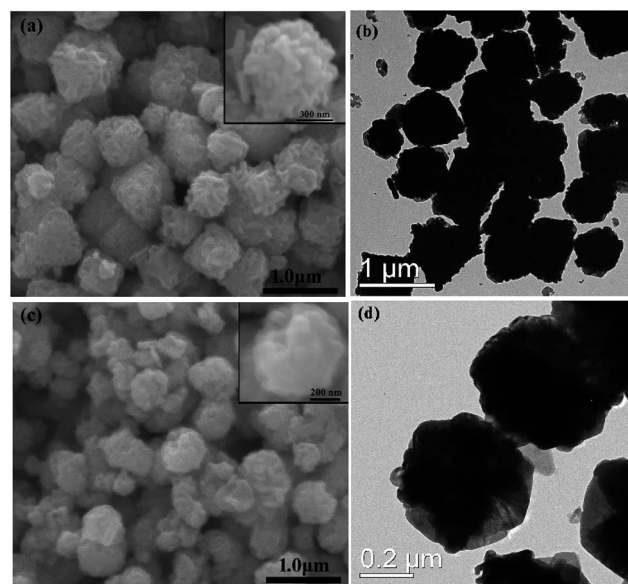


Fig. 3 FE-SEM and TEM images of Zn_2SnO_4 microspheres prepared with $V_{\text{EG}} : V_{\text{H}_2\text{O}} = 20 : 60$ (a and b) and $V_{\text{DEG}} : V_{\text{H}_2\text{O}} = 20 : 60$ (c and d), respectively, in the meantime, other experimental conditions are kept constant. Insets in (b) and (d) are the corresponding high-magnification FE-SEM images.



and DEG on the sizes of hierarchical Zn_2SnO_4 microspheres may be attributed to their coordination ability with initial Zn and Sn precursor as well as their reactivity, which can alter the Zn_2SnO_4 nuclei and growth rates on the different crystal planes, in other words, leading to different size Zn_2SnO_4 microspheres. However, further work need to be carried out to studied how EG and DEG exactly effect on the sizes of Zn_2SnO_4 microspheres.

The hierarchical Zn_2SnO_4 microsphere photoanodes are prepared by screen-printing procedures according to our previous report.¹² The photovoltaic properties of DSSCs based on three different sizes hierarchical Zn_2SnO_4 microspheres are tested by recording the J - V characteristics under AM 1.5 G illumination (100 mW cm^{-2}) conditions. The J - V curves are shown in Fig. 4a and the detailed photovoltaic parameters (J_{sc} , V_{oc} , PCEs and FF) are summarized in Table 1. The DSSC with $0.60 \mu\text{m}$ hierarchical Zn_2SnO_4 microspheres shows the lowest PCE of 2.91% with a J_{sc} of 6.17 mA cm^{-2} , a V_{oc} of 661 mV and a FF of 0.71. By contrast, the devices based on $0.85 \mu\text{m}$ hierarchical Zn_2SnO_4 microspheres shows slightly improved photovoltaic performances with a J_{sc} of 7.21 mA cm^{-2} , a V_{oc} of 691 mV, a FF of 0.71 and a PCE of 3.56%. The cells based on $1.20 \mu\text{m}$ hierarchical Zn_2SnO_4 microspheres shows the best photovoltaic performance with improved J_{sc} (7.82 mA cm^{-2}), V_{oc} (716 mV) and FF (0.71), leading to an enhanced PCE up to 4.00%. One can see a clear tendency that the J_{sc} improves with increasing sizes of the hierarchical Zn_2SnO_4 microspheres. The dark J - V characteristics in Fig. 4 also reveal that the dark photocurrents increase in the order of $1.20 \mu\text{m}$ Zn_2SnO_4 microspheres < $0.85 \mu\text{m}$ Zn_2SnO_4 microspheres < $0.60 \mu\text{m}$ Zn_2SnO_4 microspheres, indicating the suppressed charge recombination along with increased sizes of the hierarchical Zn_2SnO_4 microspheres.

The incident photon-to-current efficiency (IPCE), defined as the number of electrons generated by light in the external circuit divided by the number of incident photons, is plotted as

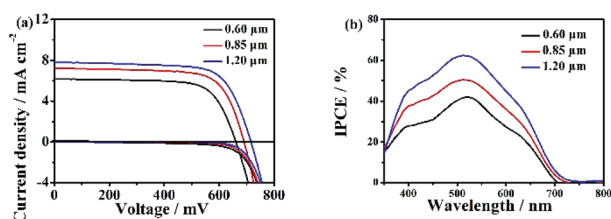


Fig. 4 (a) Photocurrent density–voltage (J - V) curves and (b) incident photon-to-current (IPCE) curves of the DSSCs devices based on different sizes Zn_2SnO_4 photoanodes with the film thickness of $\sim 15 \mu\text{m}$.

Table 1 Detailed photovoltaic parameters (J_{sc} , V_{oc} , PCE and FF) of DSSCs with different Zn_2SnO_4 photoanodes

DSSCs	$J_{\text{sc}}/\text{mA cm}^{-2}$	V_{oc}/mV	$\eta/\%$	FF	Adsorbed dye/ $\times 10^{-8} \text{ mol cm}^{-2}$
$0.60 \mu\text{m}$	6.17	661	2.91	0.71	20.16
$0.85 \mu\text{m}$	7.21	691	3.56	0.71	19.87
$1.20 \mu\text{m}$	7.82	716	4.00	0.71	18.62

a function of excitation wavelength, which shows the maximal value at 530 nm increase with the increasing microspheres size (Fig. 4b), which is consistent with the J_{sc} and η .

The loading amount of N719 dye decreases with increasing sizes of the hierarchical Zn_2SnO_4 microspheres due to the decreased surface area of the larger microspheres. In addition to the N719 dye loading amount, as well known that, the light scattering ability of the Zn_2SnO_4 photoanodes can also influence the light utilization capability through altering the path and/or extending the distance of light travelled, and in this way, the light-harvesting efficiency is improved, which would leading to an enhancement of the J_{sc} and thus PCE.

The light scattering ability of three different sizes of the hierarchical Zn_2SnO_4 microspheres films is determined by diffused reflectance spectroscopy, as shown in Fig. 5. For the larger-sized hierarchical Zn_2SnO_4 microspheres, the reflectance increases in the whole visible range (500–800 nm), indicating stronger light scattering ability which could maximize the utilization of incident light, thus enhancing the light harvesting efficiency and J_{sc} . Accordingly, the $1.20 \mu\text{m}$ Zn_2SnO_4 microspheres exhibits the best light scattering capability, which is in good agreement with its highest J_{sc} (7.82 mA cm^{-2}).

In Fig. 6, the IMPS and IMVS are performed to determine the electron transport and recombination characteristics of the devices based on the three different Zn_2SnO_4 photoanodes. The detailed parameters extracted from IMPS and IMVS measurements are summarized in Table 2 (light intensity: 150 W m^{-2}). The electron transport time (τ_d) or recombination time (τ_r) is obtained from the expression $\tau_d = 1/2\pi f_d$ or $\tau_r = 1/2\pi f_r$, respectively, where f_d or f_r is the characteristic frequency minimum of the IMPS/IMVS imaginary component. The τ_d and τ_r (Fig. 6a) decreases with increasing light intensity because the deep traps are filled by the more photo-electrons generated at higher light intensity, resulting in electron trapping/detrapping involved shallower levels.²⁰ As summarized in Table 2, the $1.20 \mu\text{m}$ Zn_2SnO_4 microspheres exhibits the shorter electron transport time (1.17 ms) than the $0.85 \mu\text{m}$ Zn_2SnO_4 microspheres (1.88 ms) and the $0.60 \mu\text{m}$ Zn_2SnO_4 microspheres (2.25 ms), implying that the $1.20 \mu\text{m}$ Zn_2SnO_4 microsphere is more efficient in terms of electron transport than smaller sized Zn_2SnO_4 microspheres. In contrast, the electron lifetime of the $1.20 \mu\text{m}$ Zn_2SnO_4 microspheres (27.76 ms) is longer than the $0.85 \mu\text{m}$ Zn_2SnO_4 microspheres (15.67 ms) and the $0.60 \mu\text{m}$ Zn_2SnO_4

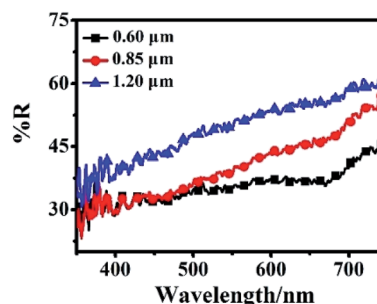


Fig. 5 Diffused reflectance spectra of different Zn_2SnO_4 films without N719 dye.



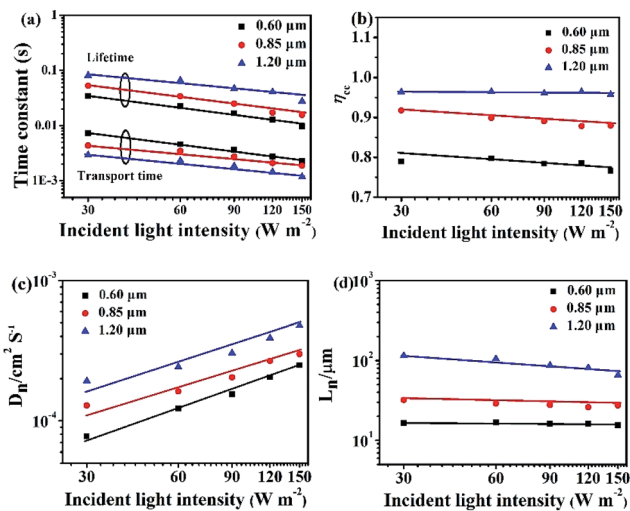


Fig. 6 (a) Light intensity dependent electron transport time and lifetime, (b) electron collection efficiency, (c) electron diffusion coefficient and (d) effective electron diffusion length of the DSSCs based on different Zn_2SnO_4 photoanodes with the film thickness of $\sim 15.0 \mu\text{m}$.

Table 2 Detailed parameters (τ_d , τ_r , η_{cc} , D_n , and L_n) extracted from IMPS and IMVS measurements for DSSCs based on different Zn_2SnO_4 photoanodes with a film thickness of $\sim 15.0 \mu\text{m}$. Light intensity: 150 W m^{-2}

DSSCs	τ_d (ms)	τ_r (ms)	$\eta_{cc}/\%$	D_n ($\text{cm}^2 \text{s}^{-1}$)	L_n (μm)
1.20 μm	1.17	27.76	95.8	4.79×10^{-4}	65.6
0.85 μm	1.88	15.67	88.0	3.00×10^{-4}	27.3
0.60 μm	2.25	9.63	76.6	2.49×10^{-4}	15.5

microspheres (9.63 ms), indicating much more suppressed charge recombination within larger sized microspheres-based devices than the small microspheres-based counterpart, thus leading to enhanced V_{oc} (716 mV) for the former.

According to the τ_d and τ_r , one can further acquire another important parameter of the DSSCs, namely, electron collection efficiency (η_{cc} , $\eta_{cc} = 1 - \tau_d/\tau_r$).^{21,22} Fig. 6b shows the η_{cc} of the DSSCs based on three different sized hierarchical Zn_2SnO_4 photoanodes. It is clear that the η_{cc} of 1.20 μm Zn_2SnO_4 microspheres-based device (95.8%) is much superior to the 0.85 μm Zn_2SnO_4 microspheres (88.0%) and 0.60 μm Zn_2SnO_4 microspheres-based counterparts (76.6%). Herein, the higher η_{cc} also contributes to the higher J_{sc} for DSSCs based on 1.20 μm Zn_2SnO_4 microspheres.

Moreover, the effective electron diffusion length (L_n) can suggest if the injected electron can transport to external circuit through the photoanode film. At the same time, the L_n can influence the J_{sc} and η . The L_n can be obtained according to the followed equation: $L_n = (D_n \times \tau_r)^{1/2}$, where $D_n = d^2/(4 \times \tau_d)$, D_n is the electron diffusion coefficient and d is the Zn_2SnO_4 film thickness. As shown in Fig. 6c and Table 2, along with the increasing spherical diameter, the D_n increases, suggesting that the electron diffusion is faster in Zn_2SnO_4 microspheres with larger size.

Based on the D_n (Fig. 6c) and τ_r (Fig. 6a), L_n can be calculated. The DSSCs based on 1.20 μm hierarchical Zn_2SnO_4 microspheres (65.6 μm) is longer than 0.85 μm hierarchical Zn_2SnO_4 microspheres (27.3 μm) and 0.60 μm hierarchical Zn_2SnO_4 microspheres (15.5 μm), implying that the photogenerated electrons can travel through a longer distance within the hierarchical Zn_2SnO_4 microspheres film ($\sim 1.20 \mu\text{m}$ in diameter), which in turn, guarantees a better charge collection efficiency for such devices.

Based on the discussion above, the enhancement of J_{sc} , V_{oc} and PCE of 1.20 μm hierarchical Zn_2SnO_4 microspheres-based DSSCs devices due to its superior light scattering, fastest electron transport rates and slowest electron recombination rates.

EIS is a powerful tool to investigate the electrochemical and photoelectrochemical kinetics processes, which can be used to measure the charge transport dynamics of the present DSSCs based on hierarchical Zn_2SnO_4 microspheres with different sizes. The Nyquist plots of the EIS are shown in Fig. 7. Typically, two semicircles are observed, which can provide good understanding and additional information on the interfacial transfer of photoexcited electrons in the present DSSCs. Generally, the first small semicircle (located in 1 kHz to 1 MHz) corresponds to the charge-transfer resistance at the interface between redox electrolyte/Pt counter electrode. One can see a similar value of resistance because of the utilization of the same electrolyte and counter electrode within the three Zn_2SnO_4 DSSCs. The second large semicircle (located in 0.1–1 kHz) is attributed to the recombination resistance across the Zn_2SnO_4 /redox electrolyte interface.¹² It is found that the recombination resistance (R_2 , obtained from Z-view software) increases gradually from 92.3 Ω to 268.6 Ω , when the sizes of hierarchical Zn_2SnO_4 microsphere size increases from 0.60 μm to 1.20 μm . And the electron lifetime of 1.20 μm Zn_2SnO_4 microspheres, 0.85 μm Zn_2SnO_4 microspheres and 0.60 μm Zn_2SnO_4 microspheres-based devices is 28.12 ms, 15.78 ms and 9.89 ms, respectively. Obviously, 1.20 μm Zn_2SnO_4 microspheres-based devices exhibit the longest electron lifetime, suggesting that the back reaction (between the injected electrons and the I_3^- in the electrolyte) is inhibited more effectively. In this case, the highest V_{oc} is

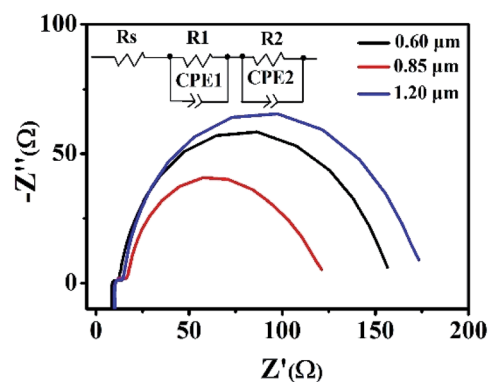


Fig. 7 The impedance spectra (Nyquist plots) of DSSCs based on the three different hierarchical Zn_2SnO_4 photoanodes with a film thickness of $\sim 15 \mu\text{m}$ measured at -0.7 V bias in the dark.



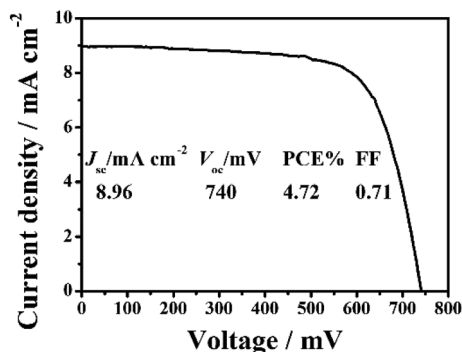


Fig. 8 J–V characteristic of DSSCs based on the ~ 15.0 μm thick hierarchical Zn_2SnO_4 microspheres (1.20 μm) coated with the TiO_2 shell.

observed. Notably, the electron lifetime variation and tendency obtained from the EIS is consistent with the aforementioned IMVS data as well.

The conduction band (CB) potential of TiO_2 is negative than Zn_2SnO_4 . $\text{Zn}_2\text{SnO}_4@ \text{TiO}_2$ core-shell structure can not only facilitate the electron transfer from TiO_2 to Zn_2SnO_4 , but also can suppress the recombination rates between the photo-injected electrons in the CB of Zn_2SnO_4 and I_3^- in the electrolyte, leading to improved η_{cc} , and thus further increasing the J_{sc} , V_{oc} and PCE.^{10,12} In this study, the coating of TiO_2 shell on Zn_2SnO_4 film is achieved *via* immersing the Zn_2SnO_4 films into TiCl_4 aqueous solution (40 mM) to form the $\text{Zn}_2\text{SnO}_4@ \text{TiO}_2$ core-shell structure (Fig. 8). As for the DSSCs based on $\text{Zn}_2\text{SnO}_4@ \text{TiO}_2$ core-shell structure, the J_{sc} increases from 7.82 mA cm^{-2} to 8.96 mA cm^{-2} , and V_{oc} increases from 716 mV to 740 mV. This results in a 25% improvement of PCE from 4.00% to 4.72%, which can be attributed to the significant reduction of the back reaction of the photo-injected electrons from Zn_2SnO_4 microspheres to the redox electrolyte (I_3^-). Another possible reason of the improvement is that defects at the internal surface can be minimized while the TiO_2 coating on the Zn_2SnO_4 microspheres surface, which facilitates the electron transfer from N719 dye molecules to the TiO_2 and the Zn_2SnO_4 microspheres. Meanwhile, the modification of TiO_2 shell layer could avoid the presence of extra internal trap sites in a poorly constructed junction on Zn_2SnO_4 microspheres.²³

4. Conclusions

In summary, a one-step hydrothermal route is demonstrated to the controllable synthesis of 3D hierarchical Zn_2SnO_4 microspheres. We proposed a plausible self-assembly formation process of Zn_2SnO_4 microspheres based on the time-dependent morphological evolution observations. When applied as photoanodes in DSSCs, the 1.20 μm Zn_2SnO_4 microspheres show the best photovoltaic performance with a PCE of 4.00%. TiCl_4 treatment results in the formation of $\text{Zn}_2\text{SnO}_4@ \text{TiO}_2$ core-shell structured photoanode, which further improve the PCE up to 4.72%. The designed 3D hierarchical Zn_2SnO_4 microspheres with promising optical and

electrical properties can be extended to other energy-related applications such as in Li-ion battery, photocatalysis and supercapacitor.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors acknowledge the financial supports from the National Natural Science Foundation of China (51502205), Application Development Foundation of Tianjin Normal University (52XK1508), Scientific Research Foundation of Tianjin Normal University (5RL131), and National Training Program of Innovation and Entrepreneurship for Undergraduates in Tianjin of China (201610065012.00) and Academic Innovation Funding of Tianjin Normal University (52XC1404).

Notes and references

- B. O'Regan and M. Grätzel, *Nature*, 1991, **353**, 737–740.
- M. Grätzel, *Nature*, 2001, **414**, 338–344.
- K. Kakiage, Y. Aoyama, T. Yano, K. Oya, J. Fujisawa and M. Hanaya, *Chem. Commun.*, 2015, **51**, 15894–15897.
- S. Colodrero, A. Mihi, L. Häggman, M. Ocaña, G. Boschloo, A. Hagfeldt and H. Míguez, *Adv. Mater.*, 2009, **21**, 764–770.
- D. H. Chen, F. Z. Huang, Y. B. Cheng and R. A. Caruso, *Adv. Mater.*, 2009, **21**, 2206–2210.
- M. Ye, D. Zheng, M. Lv, C. Chen, C. Lin and Z. Lin, *Adv. Mater.*, 2013, **25**, 3039–3044.
- W. Q. Wu, Y. F. Xu, C. Y. Su and D. B. Kuang, *Energy Environ. Sci.*, 2014, **7**, 644–649.
- Y. F. Wang, K. N. Li, C. L. Liang, Y. F. Hou, C. Y. Su and D. B. Kuang, *J. Mater. Chem.*, 2012, **22**, 21495–21501.
- Z. Dong, X. Lai, J. E. Halpert, N. Yang, L. Yi, J. Zhai, D. Wang, Z. Tang and L. Jiang, *Adv. Mater.*, 2012, **24**, 1046–1049.
- Y. F. Wang, K. N. Li, Y. F. Xu, H. S. Rao, C. Y. Su and D. B. Kuang, *Nanoscale*, 2013, **5**, 5940–5948.
- Z. Li, Y. Zhou, C. Bao, G. Xue, J. Zhang, J. Liu, T. Yu and Z. Zou, *Nanoscale*, 2012, **4**, 3490–3494.
- S. H. Choi, D. Hwang, D. Y. Kim, Y. Kervella, P. Maldivi, S. Y. Jang, R. Demadrille and I. D. Kim, *Adv. Funct. Mater.*, 2013, **23**, 3146–3155.
- T. J. Coutts, D. L. Young, X. Li, W. P. Mulligan and X. J. Wu, *J. Vac. Sci. Technol., A*, 2000, **18**, 2646–2660.
- Q. F. Zhang, C. S. Dandeneau, X. Y. Zhou and G. Z. Cao, *Adv. Mater.*, 2009, **21**, 4087–4108.
- Q. F. Zhang and G. Z. Cao, *Nano Today*, 2011, **6**, 91–109.
- S. So, I. Hwang and P. Schmuki, *Energy Environ. Sci.*, 2015, **8**, 849–854.
- J. Dou, Y. F. Li, F. Y. Xie, X. K. Ding and M. D. Wei, *Cryst. Growth Des.*, 2016, **16**, 121–125.
- Z. Q. Li, Y. P. Que, L. E. Mo, W. C. Chen, Y. Ding, Y. M. Ma, L. Jiang, L. H. Huand and S. Y. Dai, *ACS Appl. Mater. Interfaces*, 2015, **7**, 10928–10934.



- 19 Y. J. Xiong, H. G. Cai, B. J. Wiley, J. G. Wang, M. J. Kim and Y. N. Xia, *J. Am. Chem. Soc.*, 2007, **129**, 3665–3675.
- 20 T. Oekermann, T. Yoshida, H. Minoura, K. G. U. Wijayantha and L. M. Peter, *J. Phys. Chem. B*, 2004, **108**, 8364–8370.
- 21 K. Zhu, N. R. Neale, A. Miedaner and A. J. Frank, *Nano Lett.*, 2007, **7**, 69–74.
- 22 T. Oekermann, D. Zhang, T. Yoshida and H. Minoura, *J. Phys. Chem. B*, 2004, **108**, 2227–2235.
- 23 H. J. Snaith and C. Ducati, *Nano Lett.*, 2010, **10**, 1259–1265.

