

Cite this: *RSC Adv.*, 2017, 7, 40334

Electrospun magnetic CoFe₂O₄/Ag hybrid nanotubes for sensitive SERS detection and monitoring of the catalytic degradation of organic pollutants†

Wei Song,^a Zezhou Yang,^b Fuqiu Ma,^c Maoqiang Chi,^b Bing Zhao^a and Xiaofeng Lu^{*b}

We report on the facile synthesis of magnetic CoFe₂O₄/Ag hybrid nanotubes, as a reliable and sensitive surface enhanced Raman scattering (SERS) substrate for sensitive detection and *in situ* monitoring of the catalytic degradation process of organic pollutants. This SERS substrate, based on CoFe₂O₄/Ag hybrid nanotubes, is achieved through an electrospinning followed by calcination process. The CoFe₂O₄ and Ag nanoparticles are well dispersed in the CoFe₂O₄/Ag hybrid nanotubes. The unique heterostructure and strong interactions between CoFe₂O₄ and Ag nanoparticles in the hybrid nanotubes contribute the electromagnetic field SERS enhancement. In addition, target molecules can be easily enriched on the surface of the CoFe₂O₄/Ag hybrid nanotubes due to their magnetic properties, further providing good SERS properties. The CoFe₂O₄/Ag hybrid nanotubes can be also used as a catalyst for the degradation of organic pollutants. Therefore, we have developed a facile approach by using CoFe₂O₄/Ag hybrid nanotubes as both catalyst and SERS substrate to determine the reaction kinetics of the catalytic degradation of organic pollutants.

Received 15th July 2017
Accepted 14th August 2017

DOI: 10.1039/c7ra07786f

rsc.li/rsc-advances

Introduction

Surface enhanced Raman scattering (SERS) is powerful and fascinating analytical technology with a large variety of applications in biology, medical science, food security, chemical and environmental fields.^{1–6} The SERS technique can provide intrinsic structure information of target molecules with a huge enhancement (generally up to 10⁶ times) compared with traditional Raman spectroscopy.^{7–10} The most important process for SERS analysis is the fabrication of the active substrate, such as rough noble metals (*e.g.*, Au, Ag and Cu) and semiconductor nanomaterials.^{11–16} In general, the enhancement factor (EF) for rough noble metals can reach more than 10⁴ due to the existence of electromagnetic field, while the EF for semiconductors is only about 10–100, which is related to the charge-transfer complex formed between semiconductors and target molecules. To further increase the SERS activity, it is a meaningful

object to utilize both the electromagnetic enhancement of noble metals and charge-transfer enhancement of semiconductors. In the past few years, hybrid nanomaterials consisting of multicomponents have also been used as efficient SERS substrate, such as Au/TiO₂ nanotube arrays,¹⁷ Ag/TiO₂ nanofibers,¹⁸ porous CuO/Ag nanofibers,¹⁹ Ag/ZnO hollow nanospheres,²⁰ and Ag/NiO nanoflakes.²¹ In addition, the noble metal/semiconductor based hybrid nanomaterials are also good catalysts or photocatalysts, thus they can be used as SERS substrate for the *in situ* monitoring the catalytic or photocatalytic reaction process and providing the kinetic characteristic of the heterogeneous catalysis. For example, Lu, Wang and co-workers has demonstrated the fabrication of silver-coated ZnO nanowire arrays as catalyst for *in situ* monitoring of the degradation of organic pollutants in the presence of reducing agents.²² Recently, a ternary nanocomposite of ZnO–reduced graphene oxide (rGO)–Au has been prepared, which can be employed as both SERS substrate and photocatalyst for *in situ* SERS determination of typical dye molecules during the photocatalytic degradation process.²³

On the other hand, magnetic hybrid nanomaterials have attracted extensively research interest due to their multifunction in a single nanomaterial.^{24,25} In particular, the integration of noble metal nanoparticles with ferromagnetic oxides in hybrids results in a new kind of nanocatalysts with magnetically recoverable properties, which can be easily recycled from the

^aState Key Laboratory of Supramolecular Structure and Materials, Jilin University, Changchun 130012, P. R. China. E-mail: weisong@jlu.edu.cn; Fax: +86-431-85168473; Tel: +86-431-85168473

^bAlan G. MacDiarmid Institute, College of Chemistry, Jilin University, Changchun, 130012, P. R. China. E-mail: xflu@jlu.edu.cn

^cFundamental Science of Nuclear Safety and Simulation Technology Laboratory, Harbin Engineering University, 150001, P. R. China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra07786f



reaction system. For instance, the dumbbell-like Au@Fe₃O₄ heterostructures have been prepared *via* a thermal decomposition of iron-oleate complex using Au nanoparticles as the seeds.²⁶ The as-prepared Au@Fe₃O₄ heterostructure not only exhibited a high catalytic activity for the reduction of nitrophenol, but also showed an excellent magnetically recyclable properties. In addition, bifunctional Au-Fe₃O₄ hybrid hollow spheres have also been synthesized through a one-pot hydrothermal reaction, displaying both good catalytic activity and efficient SERS sensitivity.²⁷ Owing to the enrichment of target molecules through magnetism-induced aggregation, silver-coated magnetic nanoparticles have been proved to be versatile SERS substrate with a high sensitivity.²⁸ Importantly, the integrating of Au nanoparticles with Fe₃O₄/C nanoparticles produces a SERS substrate with good catalytic hydrogenation property, which offers a platform for *in situ* SERS monitoring of the reduction of *p*-nitrothiophenol to *p*-aminothiophenol.²⁹

Owing to the high coercive force, good saturation magnetization, and chemical stability, CoFe₂O₄ nanomaterial have become an alternative to Fe₃O₄ to support noble metal nanoparticles for catalytic applications in the past few years.^{30–32} In this regard, we have prepared rGO supported CoFe₂O₄-Pd nanoparticles *via* a one-pot microwave synthetic route, demonstrating a high catalytic activity as well as excellent magnetic recoverable properties.³² However, up to now, there are few reports on the fabrication of CoFe₂O₄ based hybrid nanomaterials as both SERS substrate and nanocatalyst for *in situ* monitoring of the heterogeneous catalytic reaction process. In addition, CoFe₂O₄ is also a p-type semiconductor with a narrow band gap (0.9 eV).³³ When CoFe₂O₄ is combined with Ag nanoparticles, a charge transfer from CoFe₂O₄ to Ag nanoparticles will induce a large electromagnetic field, which improve the SERS activities.

In this work, we report a simple electrospinning combined with calcination process to prepare CoFe₂O₄/Ag hybrid nanotubes as SERS substrate. The synthesized CoFe₂O₄/Ag hybrid nanotubes show a uniform morphology and good magnetic property, thus they can be quickly aggregated in a small area in the solution to enrich target molecules and to accelerate the SERS detection rate. And the SERS activity has been significantly increased due to the electromagnetic field contributed by the strong interactions between CoFe₂O₄ and Ag nanoparticles in the hybrid nanotubes. Furthermore, the prepared CoFe₂O₄/Ag hybrid nanotubes are good catalysts for the degradation of typical organic pollutant (methylene blue, MB), which can be *in situ* monitored by the SERS spectra. The experimental results display that the synthesized CoFe₂O₄/Ag hybrid nanotubes possess high catalytic activity and the catalytic reaction follows pseudo-first-order kinetics.

Experimental

Materials

Poly(vinylpyrrolidone) (PVP, $M_w = 1\,300\,000$) and *p*-aminothiophenol (PATP) were purchased from Sigma-Aldrich. Fe(NO₃)₃·9H₂O was obtained from Tianjin East China reagent factory. Co(Ac)₂·4H₂O and NaBH₄ were purchased from

Sinopharm chemical reagent Beijing Co., Ltd. AgNO₃ and ethanol was bought from Beijing Chemical Works. MB was purchased from Tianjin Guangfu Fine Chemical Research Institute. *N,N'*-Dimethylformamide (DMF) was commercially obtained from Tianjin Tiantai Fine Chemicals Co., Ltd. All the chemicals were used as received without purification. Deionized water was used throughout the study.

Preparation of CoFe₂O₄/Ag hybrid nanotubes *via* an electrospinning combined with calcination process

In a typical procedure, 0.656 g of PVP was dissolved in a mixing solvent consisting of 4.56 g of ethanol and 3.12 g of DMF. Then 0.0616 g of Co(Ac)₂·4H₂O together with 0.20 g of Fe(NO₃)₃·9H₂O and 0.042 g of AgNO₃ was added into the above solution. The mixture was stirred for more than 12 h at room temperature to produce a homogeneous solution. The as-prepared mixing solution was electrospun under an applied high voltage of 15 kV through a high-voltage DC power supply. A piece of aluminum foil was used as a collector for collecting the nanofibers. The distance between the spin nozzle and the collector was about 20 cm. The as-prepared fibrous membrane was calcined at 550 °C in air for 3 h subsequently according to a confirmed program. Finally, CoFe₂O₄/Ag hybrid nanotubes membrane with a black colour were obtained. For a comparison, individual electrospun CoFe₂O₄ nanotubes and Ag nanomaterials have also been prepared *via* a similar electrospinning followed by calcination process.

Catalytic measurements

In a typical procedure, 30 µL of MB solution (10^{−3} M) was added into 3 mL of deionized water. Then 50 µL of NaBH₄ solution (30 mg mL^{−1}) was added. After that, 100 µL of CoFe₂O₄/Ag hybrid nanotubes aqueous dispersions (3 mg mL^{−1}) was added into the above solution. UV-vis spectral measurement was used to evaluate the catalytic reaction.

SERS measurements

In a typical experiment, PATP was used as a Raman probe to estimate the response of the CoFe₂O₄/Ag hybrid nanotube substrate for SERS measurements. 20 µL of CoFe₂O₄/Ag hybrid nanotubes dispersions (1 mg mL^{−1}) was mixed with 180 µL aqueous solution of PATP with a certain concentration for 5 min under sonication. After standing for about 30 min, the CoFe₂O₄/Ag hybrid nanotubes were separated from the solution by using an external magnet, and the precipitate was dropped onto a clean glass slide. The air dried precipitate was then analysed with a Raman spectrometer. PATP samples were prepared in aqueous solutions with concentrations of 10^{−5}, 10^{−6}, 10^{−7}, 10^{−8}, 10^{−9} M.

SERS monitoring of the catalytic degradation of MB on the CoFe₂O₄/Ag hybrid nanotube

In the procedure for SERS monitoring of the catalytic degradation of MB, 0.1 mL of CoFe₂O₄/Ag hybrid nanotube dispersions (1 mg mL^{−1}) was mixed with 0.9 mL of MB solution (10^{−5} M).



Then 5 μL of 2.5×10^{-2} M NaBH_4 aqueous solution was added into 55 μL of the above mixture solution. The catalytic degradation of MB was measured by monitoring the SERS spectra of the final product at different reaction time.

Characterization

The morphologies of the synthesized $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes were characterized by field-emission scanning electron microscopy (SEM, FEI Nova NanoSEM 450) and transmission electron microscopy (TEM, JEOLJEM-1200 EX) operated at 15 and 100 kV, respectively. HRTEM imaging with Energy dispersive X-ray (EDX) and elemental mapping analysis was performed with a FEI Tecnai G2 F20 high resolution transmission electron microscope operated at a 200 kV accelerating voltage. X-ray data were collected by using an X-ray diffractometer (Empyrean, PANalytical B.V.) based on Cu-K α radiation. Analysis of the X-ray photoelectron spectra (XPS) was performed on a thermo ESCALAB 250 spectrometer. The ultraviolet-visible (UV-vis) absorption spectroscopy was performed on a Shimadzu UV-2501 PC spectrometer. SERS spectra of PATP and the catalytic degradation of MB were measured with a Renishaw-1000 spectrometer with a He/Ne laser as excitation line of 532 nm.

Results and discussion

$\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes are successfully prepared through a two-step fabrication process (Fig. 1). First, electrospun PVP/ $\text{Co}(\text{Ac})_2/\text{Fe}(\text{NO}_3)_3/\text{AgNO}_3$ composite nanofibers membrane has been prepared *via* an electrospinning technique. Subsequently, the as-prepared composite nanofibers are calcined in air at 550 $^\circ\text{C}$ to produce $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. Accordingly, the color turns from white to black after the calcination, indicating the decomposition of PVP. Fig. 2a shows the SEM image of the as-electrospun PVP/ $\text{Co}(\text{Ac})_2/\text{Fe}(\text{NO}_3)_3/\text{AgNO}_3$ composite nanofibers. It has been obviously seen that the products are nanofibers with diameters from 100 to 400 nm. After calcination, the size of the nanofibers becomes thinner and their diameters are in the range of 50–150 nm (Fig. 2b and c). In addition, the TEM image is also used to characterize the morphology of the calcined products, revealing a hollow characteristic of the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanostructures. As shown in Fig. 2d, the TEM also shows that the surface of $\text{CoFe}_2\text{O}_4/\text{Ag}$

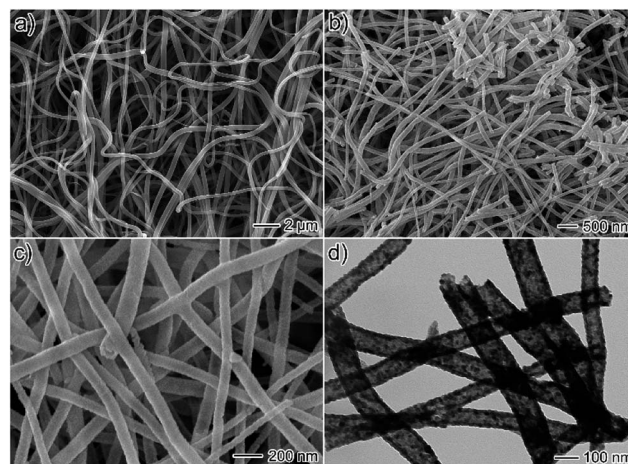


Fig. 2 (a) SEM image of the electrospun PVP/ $\text{Co}(\text{Ac})_2/\text{Fe}(\text{NO}_3)_3/\text{AgNO}_3$ composite nanofibers; (b and c) SEM images of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes with a low and high magnifications; (d) TEM images of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes.

hybrid nanotubes is composed of many small nanoparticles and the thickness of the wall of the hybrid nanotubes is about 15 nm.

Fig. 3a shows a single $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube and the selected area electron diffraction (SAED) pattern of the product, demonstrating their high crystallinity. The size of the CoFe_2O_4 and Ag nanoparticles in $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes is from several to tens of nanometers. And these nanoparticles almost inosculate as a whole shell in the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. The SAED pattern of the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes demonstrates the existence of the crystal CoFe_2O_4 and Ag nanoparticles (inset in Fig. 3a). From Fig. 3b, CoFe_2O_4 with a lattice spacing of 0.48 nm corresponding to the (111) lattice plane of cubic CoFe_2O_4 with a spinel structure is clearly

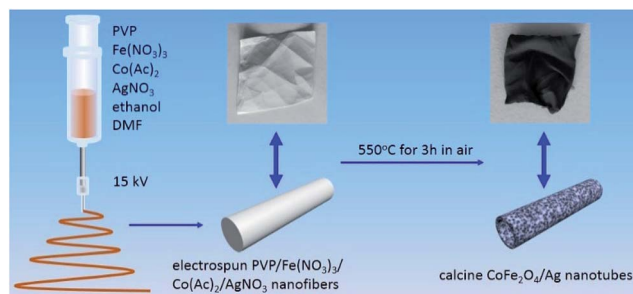


Fig. 1 Schematic illustration of the fabrication of the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes through electrospinning followed by a calcination process.

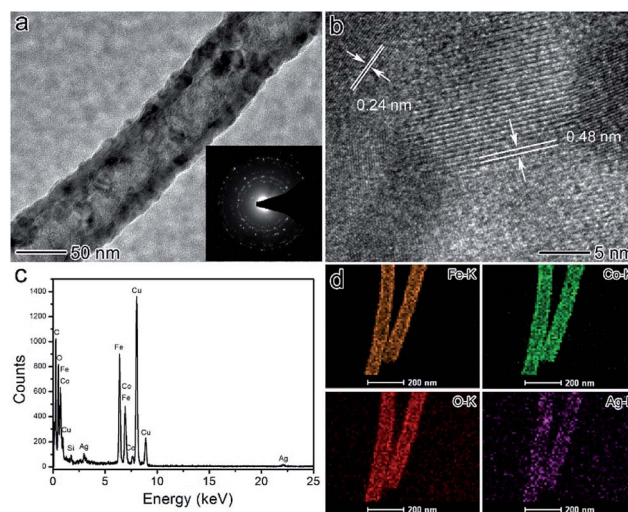


Fig. 3 (a) TEM image of a single $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube, the inset shows the corresponding ED pattern; (b) HRTEM image of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube; (c and d) EDX spectrum and EDX mapping of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes.



observed in the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. In addition, Ag is well crystallized exhibiting a clear lattice with a spacing of 0.22 nm, which is attributed to the (111) lattice plane of face-centered cubic (fcc) of Ag. The as-synthesized $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes are also characterized by energy-dispersive X-ray (EDX) spectroscopy (Fig. 3c), which displays the existence of C, O, Fe, Co, Ag, Cu, Si and no other obvious elements are observed. The Cu and Si elements originate from carbon coated copper grid and the instrument substrate, respectively. This result demonstrates the successful formation of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. The elemental mapping in Fig. 3d demonstrates the direct evidence that Fe, Co, O and Ag elements are visible throughout the whole shell of hybrid nanotubes, indicating that the hybrid nanotubes are composed of CoFe_2O_4 and Ag nanoparticles.

FTIR spectroscopy is employed to examine the removal of PVP after the calcination. As shown in the FTIR spectrum of electrospun $\text{PVP}/\text{Co}(\text{Ac})_2/\text{Fe}(\text{NO}_3)_3/\text{AgNO}_3$ hybrid nanofibers in Fig. 4a, dominant and distinct peak of PVP at 1674 cm^{-1} is ascribed to the $\text{C}=\text{O}$ stretching vibration, while the peaks at 1289 and 651 cm^{-1} are related to the $\text{C}-\text{O}-\text{C}$ stretching vibration, and the peak at 2923 and 1493 cm^{-1} can be assigned of CH_2 absorption, the peak at 1374 cm^{-1} is due to the aliphatic CH group vibration.³⁴ In addition, the peak at 3500 cm^{-1} can be attributed to the symmetric vibration of $-\text{OH}$ group of the residue water. However, the typical bands corresponding to the PVP molecules almost disappear after calcination at 550°C . Meanwhile, a new peak at 586 cm^{-1} attributed to stretching vibrations of tetrahedral complexes is clearly observed, indicating the decomposition of PVP and the formation of CoFe_2O_4 .³⁵ To examine the crystallographic features of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes, XRD is performed on the product. With regard to the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes, all the diffraction peaks are matched well with crystal CoFe_2O_4 and Ag. The typical peaks at 30.3 , 35.5 , 43.1 , 54.4 , 57.2 , 62.8 , and 74.1° are assigned to the (220), (311), (400), (422), (511), (440) and (533) planes of CoFe_2O_4 (JCPDS 22-1086). While the peaks at 38.2 , 44.4 , 64.5 , and 77.5° are attributed to (111), (200), (220), (311) planes of face centered cubic Ag phase (JCPDS No. 04-0783) (Fig. 4b). All the diffraction peaks are sharp and intense, demonstrating the high crystallinity of the CoFe_2O_4 and Ag phase in the hybrid nanotubes. We have also estimated the size of Ag nanoparticles based on the Scherrer equation ($D_{hkl} = \kappa\lambda \times$

$57.3/\beta \cos \theta$), where D is the average diameter of the Ag nanoparticles, κ is the shape factor ($\kappa = 0.89$), λ is the X-ray wavelength ($\lambda = 1.5418\text{ \AA}$), β is the line broadening measured as the half-height in radians, θ is the Bragg angle in degrees. From the XRD pattern of Ag in Fig. 4b, the average size of Ag nanoparticles is calculated to be about 32.4 nm .

Further information on the chemical composition and valence state of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube are also attained by the X-ray photoelectron spectroscopy (XPS) measurement. The survey spectrum clearly displays C, O, Fe, Co, and Ag elements in the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube product (Fig. 5a), which is in agreement with the EDX results. The XPS spectrum of Ag element shows two distinct binding energies at 368.4 and 374.4 eV , which are ascribed to the signals for $\text{Ag } 3d_{5/2}$ and $\text{Ag } 3d_{3/2}$, suggesting the formation of metallic Ag (Fig. 5b).³⁶ From Fig. 5c, it can be seen that four predominant peaks are observed, which of the first two peaks at around 781.3 and 796.7 eV are assigned to $\text{Co } 2p_{3/2}$ and $\text{Co } 2p_{1/2}$, and the peaks with higher binding energies at about 787.3 and 805.2 eV are attributed to the shake-up satellites.³⁷ In Fe 2p XPS spectrum in Fig. 5d, we can effortlessly observe two characteristic peaks of Fe $2p_{3/2}$ and Fe $2p_{1/2}$ at around 711.8 and 725.3 eV .³⁷ In addition, there are two types of oxygen species in the XPS spectrum of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes (Fig. 5e). The fitting band with a binding energy at around 530.2 eV is related to the lattice oxygen, and the fitting peak at 532.2 eV belongs to hydroxyl group, indicating that the surface of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes is hydroxylated to some extent.³⁷

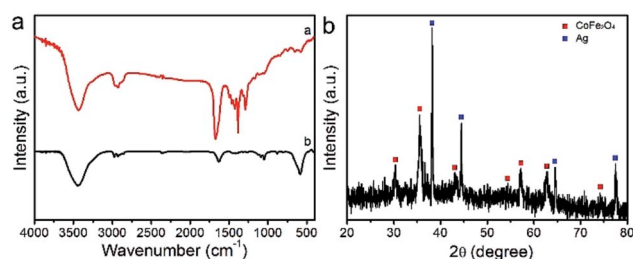


Fig. 4 (a) FTIR spectra of the electrospun $\text{PVP}/\text{Co}(\text{Ac})_2/\text{Fe}(\text{NO}_3)_3/\text{AgNO}_3$ composite nanofibers and the calcined $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes; (b) XRD pattern of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes.

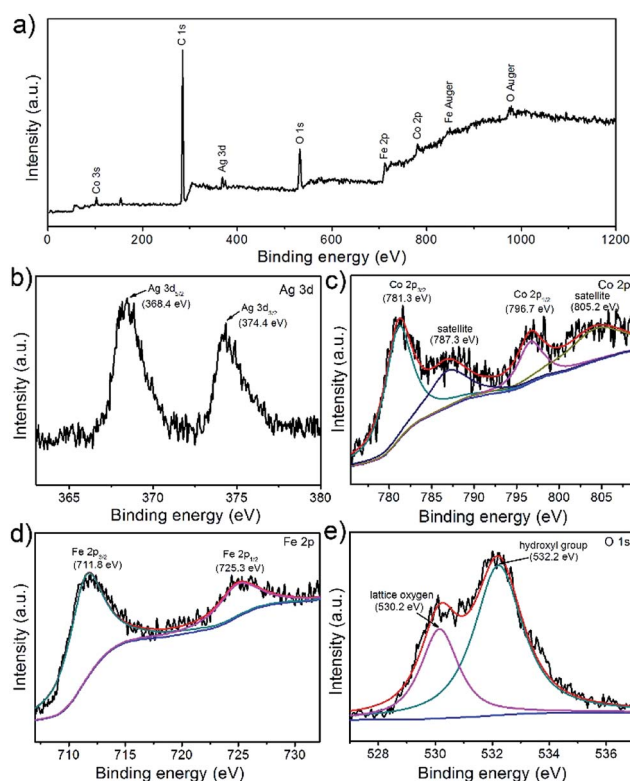


Fig. 5 XPS spectra of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes: (a) full survey; (b) Ag 3d; (c) Co 2p; (d) Fe 2p; (e) O 1s.



It is highly desirable for the rapid and sensitive detection and analysis of some small organic molecules for biotechnology and environment science. Among a large variety of strategies for quantitative analysis of organic molecules, SERS is a great powerful and sensitive spectroscopic technique to characterize the chemical structure of the adsorbed target molecules. In this study, PATP is employed as a model probe molecule which is adsorbed on the surface of CoFe₂O₄/Ag hybrid nanotubes. As shown in Fig. 6, PATP molecules can adsorb on the surface of CoFe₂O₄/Ag hybrid nanotubes in aqueous solution. Then after the magnetic enrichment, PATP adsorbed CoFe₂O₄/Ag hybrid nanotubes is collected for SERS measurement. Fig. 6a shows the SERS spectra of PATP molecules with varied concentrations from 10⁻⁵ to 10⁻⁹ M on the surface of CoFe₂O₄/Ag hybrid nanotubes. It is found that strong Raman bands is observed when the concentration of PATP is 10⁻⁵ M. The characteristic peaks at 1574 and 1072 cm⁻¹ can be attributed to the a₁ modes of PATP, and the first peak is due to the C–C stretching, while the other peak is related to the C–S stretching mode.³⁶ These two peaks is shifting to lower wavenumbers compared to the Raman spectrum of PATP molecules, which should be due to the strong interactions between PATP molecules and the CoFe₂O₄/Ag hybrid nanotubes substrate.³⁶ On the other hand, some other predominant bands at 1434, 1387, 1186, 1139 cm⁻¹ have also been clearly observed. These bands ascribed to b₂ symmetry of PATP molecules exhibit strong resonance enhancement owing to the charge transfer.^{36,38} It is also found that all the SERS bands of PATP molecules are strongly dependent on their concentrations. As shown in Fig. 6b, the SERS intensities of the typical a₁ and b₂ bands of PATP molecules decrease with the decreasing of their concentrations. However, the characteristic bands of PATP molecules can still be distinguished with a concentration down to 10⁻⁹ M, demonstrating the high SERS sensitivity of CoFe₂O₄/Ag hybrid nanotubes. The excellent SERS properties may be owing to the large electromagnetic field induced by the charge transfer from CoFe₂O₄ to Ag nanoparticles. It is well known that the detection limit is very important for the SERS substrate, thus we have compared the detection limit of the as-prepared CoFe₂O₄/Ag hybrid SERS substrate with the previous reports. It is found that the

Table 1 The comparison of the SERS property on the different types of substrates using PATP molecules as probes

SERS substrate	Limit of detection (M)	References
Pd nanosphere	10 ⁻⁶	39
Ag nanoparticles	10 ⁻⁷	40
Au nanoparticles-functionalized monolithic columns	10 ⁻⁷	41
SiO ₂ -isolated Ag islands	10 ⁻⁸	42
Nanoporous Ag microstructure	10 ⁻⁸	43
Ag nanotriangles-loaded filter paper	10 ⁻⁸	44
Ag@carbon dots hybrid	10 ⁻⁸	45
Three-dimensional silver nanoparticles decorated plasmonic paper	10 ⁻⁹	46
Au-coated MnFe ₂ O ₄ magnetic nanoparticles	10 ⁻⁹	47
Electrospun CoFe ₂ O ₄ /Ag hybrid nanofibers	10 ⁻⁹	This work

detection limit of this work is lower or comparable with most of the previous reported SERS substrates, such as Pd nanospheres,³⁹ Ag nanoparticles,⁴⁰ Au nanoparticles-functionalized monolithic columns,⁴¹ SiO₂-isolated Ag islands,⁴² nanoporous Ag microstructure,⁴³ Ag nanotriangles-loaded filter paper,⁴⁴ Ag@carbon dots hybrid,⁴⁵ three-dimensional Ag nanoparticles decorated plasmonic paper,⁴⁶ Au-coated MnFe₂O₄ magnetic nanoparticles,⁴⁷ demonstrating an efficient SERS property of the as-prepared CoFe₂O₄/Ag hybrid nanotubes (Table 1).

The finite difference time domain (FDTD) has been performed to elucidate the mechanism of the SERS enhancement on CoFe₂O₄/Ag hybrid SERS substrate. As shown in Fig. 7a and b, the higher field enhancement is localized at the gap between CoFe₂O₄ and Ag components in the hybrid nanotubes. In general, when semiconducting CoFe₂O₄ are combined with Ag nanoparticles, a charge transfer from CoFe₂O₄ to Ag nanoparticles usually takes place due to their different work functions. Thus an electromagnetic field will be generated at the gap between CoFe₂O₄ and Ag surface, resulting in a SERS enhancement. This result has been proved by the FDTD results.

The synthesized CoFe₂O₄/Ag hybrid nanotubes are not only an efficient SERS substrate, but also a good catalyst for the

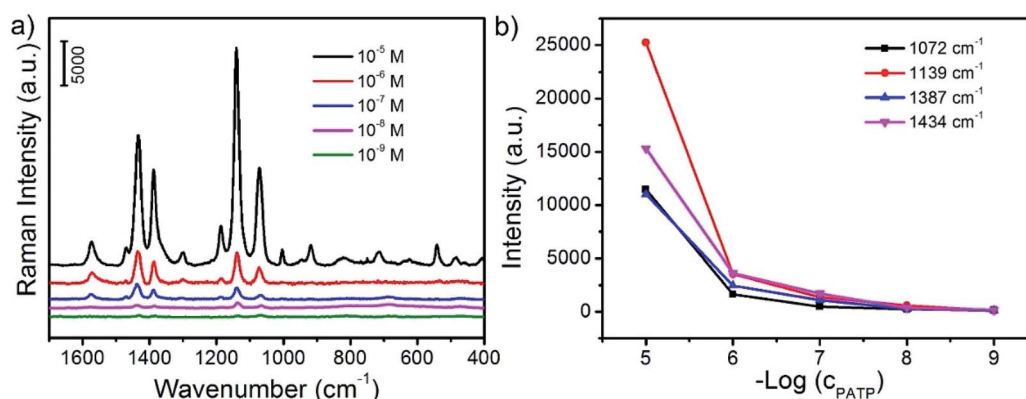


Fig. 6 (a) SERS spectra of PATP molecules with different concentrations from 10⁻⁵ to 10⁻⁹ M on the surface of the prepared CoFe₂O₄/Ag hybrid nanotubes; (b) the relationship between the intensity of the band peaking at 1072, 1139, 1387 and 1434 cm⁻¹ and -log (c_{PATP}).



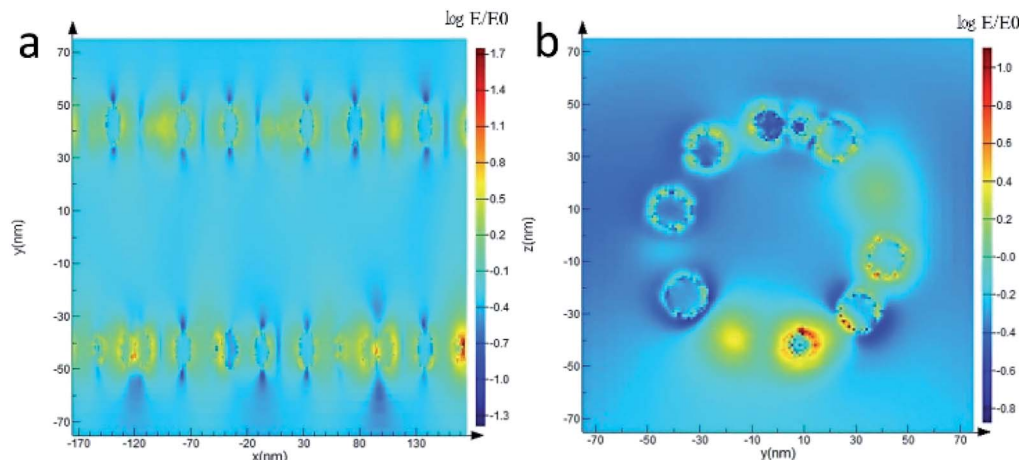


Fig. 7 The distribution of the electric field for $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube substrate calculated with FDTD simulation. (a) Longitudinal section, (b) cross section.

degradation of organic pollutant. In this work, we have studied the catalytic activity of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes by employing the degradation of MB in the presence of excess amount of NaBH_4 . The degradation process can be monitored by the UV-vis absorption spectra. As shown in Fig. S1a,[†] the MB aqueous solution exhibits an obvious absorption peak at around 662 nm, while the intensity of this peak become weaker and weaker after the addition of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes and almost disappears in 5 min. This result indicates that the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes are good catalyst toward the degradation of MB by NaBH_4 . From Fig. S1b,[†] the rate constant k can be calculated from the rate equation and the value is 0.396 min^{-1} . We have also compared the catalytic activity of the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes with that of individual CoFe_2O_4 nanotubes and electrospun Ag nanomaterials. It is found that individual CoFe_2O_4 nanotubes and electrospun Ag nanomaterials almost do not show catalytic activity for the degradation of MB in the presence of NaBH_4

(Fig. S2[†]). This result demonstrates the synergistic effect between CoFe_2O_4 and Ag components in the hybrid nanotubes for the degradation of MB.

Due to the ability of dual applications as both catalyst and SERS substrate of the prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes, an *in situ* SERS monitoring of the catalytic degradation of MB has been developed. SERS monitoring of the degradation of MB is performed in a liquid system in the presence of NaBH_4 using a 532 nm laser excitation, and the SERS signals are directly collected from the surface of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. Fig. 8a shows the SERS spectra of the MB molecules in the presence of NaBH_4 by using $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes as SERS substrate and catalyst at different time intervals. It is evident that three predominant bands at 1623, 1425, and 1394 cm^{-1} appear in the SERS spectrum of MB before the addition of NaBH_4 in the solution system. These typical bands can be assigned to C–C ring stretching, C–N stretching and N–C–H in-plane bending vibrations.⁴⁸ During the catalytic reaction

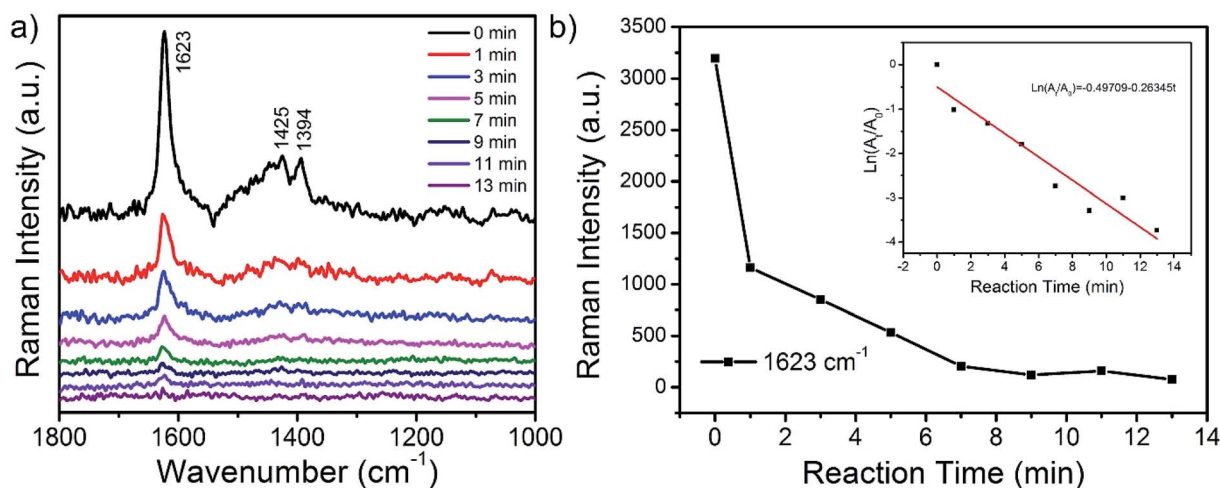


Fig. 8 (a) The change of SERS spectra for the catalytic degradation of MB on the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube substrate. (b) The curve of the relationship between the Raman intensity at the band of 1623 cm^{-1} and the reaction time. The inset figure shows the linear relationship between $\ln(A_t/A_0)$ and the reaction time according to the bands at 1623 cm^{-1} .



process in the presence of NaBH_4 , the SERS intensity of the typical bands of MB molecules gradually decreases with increasing time and the characteristic peaks of MB almost disappear in 13 min, revealing the degradation of MB molecules on the surface of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. The degradation of MB by $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes may be explained by a Langmuir–Hinshelwood model. Firstly, MB was adsorbed on the surface of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. Then NaBH_4 donates electrons to Ag in $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. Finally, the electrons transfer from Ag nanoparticles to MB, leading to the reduction of MB on the surface of $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes. The catalytic reaction kinetics of the degradation of MB is also studied. Fig. 8b shows the relationship between $\ln(A_t/A_0)$ and the reaction time for the degradation of MB catalyzed by $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes, wherein A_t and A_0 stand for the SERS intensities at 1623 cm^{-1} at time t and 0, respectively. Accordingly, the inset of Fig. 8b exhibits a good linear relationship between logarithmic integrated intensity and reaction time, and the rate law can be expressed as a fitting equation of $\ln(A_t/A_0) = -0.49709 - 0.26345t$. Based on the fitting equation, it can be concluded that the kinetics of the catalytic degradation of MB follows the pseudo-first order reaction. The catalytic reaction rate constant is calculated to be about 0.26 min^{-1} .

Conclusions

In summary, we have prepared a magnetic $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotube as SERS substrate *via* an electrospinning followed by a calcination process. The unique heterostructure and magnetic enrichment of target molecules contribute a large SERS enhancement on this substrate. The $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes are also good catalysts for the degradation of organic pollutants. By combining the catalytic and SERS properties of the $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes towards the degradation of MB, a simple approach for the *in situ* monitoring of the catalytic reaction and its kinetics using SERS technique has been developed. Owing to the unique SERS properties and excellent catalytic activity, the as-prepared $\text{CoFe}_2\text{O}_4/\text{Ag}$ hybrid nanotubes show great promising potential applications in the fields of catalysis, biosensing and environmental monitoring.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the research grants from the National Natural Science Foundation of China (21473068, 51473065, 21327803), Jilin Science and Technology Department Project (20150101034JC) and the Natural Science Foundation of Heilongjiang Province (B201317). We thank Prof. Shuping Xu and Mr Yu Tian for their help with the FDTD simulation.

Notes and references

- 1 S. Lal, N. K. Grady, J. Kundu, C. S. Levin, J. B. Lassiter and N. J. Halas, *Chem. Soc. Rev.*, 2008, **37**, 898–911.
- 2 L. Tong, T. Zhu and Z. Liu, *Chem. Soc. Rev.*, 2011, **40**, 1296–1304.
- 3 X. M. Qian and S. M. Nie, *Chem. Soc. Rev.*, 2008, **37**, 912–920.
- 4 K. Kneipp, M. Moskovits and H. Kneipp, *Surface-Enhanced Raman Scattering-Physics and Applications*, Springer, Heidelberg and Berlin, 2006.
- 5 E. Le Ru and P. Etchegoin, *Principles of Surface-Enhanced Raman Spectroscopy and Related Plasmonic Effects*, Elsevier, Amsterdam, 2009.
- 6 Y. Ozaki, K. Kneipp and R. Aroca, *Frontiers of Surface-Enhanced Raman Scattering: Single Nanoparticles and Single Cells*, John Wiley & Sons, Inc., 2014.
- 7 L. Guerrini and D. Graham, *Chem. Soc. Rev.*, 2012, **41**, 7085–7107.
- 8 B. Sharma, R. R. Frontiera, A. I. Henry, E. Ringe and R. P. VanDuyne, *Mater. Today*, 2012, **15**, 16–25.
- 9 D. Y. Wu, J. F. Li, B. Ren and Z. Q. Tian, *Chem. Soc. Rev.*, 2008, **37**, 1025–1041.
- 10 S. E. J. Bell and N. M. S. Sirimuthu, *Chem. Soc. Rev.*, 2008, **37**, 1012–1024.
- 11 H. Y. Liang, Z. P. Li, W. Z. Wang, Y. S. Wu and H. X. Xu, *Adv. Mater.*, 2009, **21**, 4614–4618.
- 12 Y. Lu, G. L. Liu and L. P. Lee, *Nano Lett.*, 2005, **5**, 5–9.
- 13 M. V. Canameres, J. V. Garcia-Ramos, J. D. Gomez-Varga, C. Domingo and S. Sanchez-Cortes, *Langmuir*, 2005, **21**, 8546–8553.
- 14 X. X. Han, W. Ji, B. Zhao and Y. Ozaki, *Nanoscale*, 2017, **9**, 4847–4861.
- 15 D. Maznichenko, K. Venkatakrishnan and B. Tan, *J. Phys. Chem. C*, 2013, **117**, 578–583.
- 16 W. Li, R. Zamani, P. Rivera Gil, B. Pelaz, M. Ibáñez, D. Cadavid, A. Shavel, R. A. Alvarez-Puebla, W. J. Parak, J. Arbiol and A. Cabot, *J. Am. Chem. Soc.*, 2013, **135**, 7098–7101.
- 17 X. Li, G. Chen, L. Yang, Z. Jin and J. Liu, *Adv. Funct. Mater.*, 2010, **20**, 2815–2824.
- 18 W. Song, Y. X. Wang and B. Zhao, *J. Phys. Chem. C*, 2007, **111**, 12786–12791.
- 19 W. Wang, Z. Y. Feng, W. Jiang and J. H. Zhan, *CrystEngComm*, 2013, **15**, 1339–1344.
- 20 X. He, C. Yue, Y. Zang, J. Yin, S. Sun, J. Li and J. Kang, *J. Mater. Chem. A*, 2013, **1**, 15010–15015.
- 21 Q. Zhou, G. Meng, Q. Huang, C. Zhu, H. Tang, Y. Qian, B. Chen and B. Chen, *Phys. Chem. Chem. Phys.*, 2014, **16**, 3686–3692.
- 22 X. M. Zhao, B. H. Zhang, K. L. Ai, G. Zhang, L. Y. Cao, X. J. Liu, H. M. Sun, H. S. Wang and L. H. Lu, *J. Mater. Chem.*, 2009, **19**, 5547–5553.
- 23 C. Y. Wen, F. Liao, S. S. Liu, Y. Zhao, Z. H. Kang, X. L. Zhang and M. W. Shao, *Chem. Commun.*, 2013, **49**, 3049–3051.
- 24 D. Wang and D. Astruc, *Chem. Rev.*, 2014, **114**, 6949–6985.
- 25 V. Polshettiwar, R. Luque, A. Fihri, H. B. Zhu, M. Bouhrara and J. M. Basset, *Chem. Rev.*, 2011, **111**, 3036–3075.



- 26 F. H. Lin and R. A. Doong, *J. Phys. Chem. C*, 2011, **115**, 6591–6598.
- 27 Q. Gao, A. W. Zhao, H. Y. Guo, X. C. Chen, Z. B. Gan, W. Y. Tao, M. F. Zhang, R. Wu and Z. X. Li, *Dalton Trans.*, 2014, **43**, 7998–8006.
- 28 Y. F. Pang, R. Xiao and S. Q. Wang, *Colloids Surf., A*, 2016, **506**, 393–401.
- 29 W. Y. Cai, X. H. Tang, B. Sun and L. B. Yang, *Nanoscale*, 2014, **6**, 7954–7958.
- 30 G. Hu, J. H. Choi, C. B. Eom, V. G. Harris and Y. Suzuki, *Phys. Rev. B: Condens. Matter*, 2000, **62**, R779–R782.
- 31 W. N. Sun, X. F. Lu, Y. P. Xue, Y. Tong and C. Wang, *Macromol. Mater. Eng.*, 2014, **299**, 361–367.
- 32 X. F. Lu, L. Yang, X. J. Bian, D. M. Chao and C. Wang, *Part. Part. Syst. Charact.*, 2014, **31**, 245–251.
- 33 A. V. Ramos, M. J. Guittet, J. B. Moussy, R. Mattana, C. Deranlot, F. Petroff and C. Gatel, *Appl. Phys. Lett.*, 2007, **91**, 122107.
- 34 P. J. Yao, J. Wang, Q. Qiao and H. Y. Du, *J. Mater. Sci.*, 2015, **50**, 1338–1349.
- 35 M. P. Reddy, A. M. A. Mohamed, X. B. Zhou, S. Du and Q. Huang, *J. Magn. Magn. Mater.*, 2015, **388**, 40–44.
- 36 W. Song, W. Ji, S. Vantasin, I. Tanabe, B. Zhao and Y. Ozaki, *J. Mater. Chem. A*, 2015, **3**, 13556–13562.
- 37 Z. Z. Yang, Z. Zhang, Y. Z. Jiang, M. Q. Chi, G. D. Nie, X. F. Lu and C. Wang, *RSC Adv.*, 2016, **6**, 33636–33642.
- 38 Z. Mao, W. Song, X. X. Xue, W. Ji, Z. S. Li, L. Chen, H. J. Mao, H. M. Lv, X. Wang, J. R. Lombardi and B. Zhao, *J. Phys. Chem. C*, 2012, **116**, 14701–14710.
- 39 L. M. Wang, L. H. Wang, E. Z. Tan, L. Guo and X. D. Han, *Nanotechnology*, 2011, **22**, 305712.
- 40 T. H. D. Nguyen, P. Zhou, A. Mustapha and M. Lin, *Analyst*, 2016, **141**, 5382–5389.
- 41 Q. Jiang, T. Zeng, S. Yang, Q. Chen, L. Chen, Y. Ye, J. Zhou and S. P. Xu, *Spectrochim. Acta, Part A*, 2015, **141**, 244–251.
- 42 Y. X. Wang, X. Y. Zhao, L. Chen, S. Chen, M. B. Wei, M. Gao, Y. Zhao, C. Wang, X. Qu, Y. J. Zhang and J. H. Yang, *Langmuir*, 2014, **30**, 15285–15291.
- 43 K. WONGravee, H. Gatemala, C. Thammacharoen, S. Ekgasit, S. Vantasin, I. Tanabe and Y. Ozaki, *RSC Adv.*, 2015, **5**, 1391–1397.
- 44 C. Wang, B. Liu and X. Dou, *Sens. Actuators, B*, 2016, **231**, 357–364.
- 45 J. Jin, S. Zhu, Y. Song, H. Zhao, Z. Zhang, Y. Guo, J. Li, W. Song, B. Yang and B. Zhao, *ACS Appl. Mater. Interfaces*, 2016, **8**, 27956–27965.
- 46 Y. X. Li, K. Zhang, J. J. Zhao, J. Ji, C. Ji and B. H. Liu, *Talanta*, 2016, **147**, 493–500.
- 47 J. F. Wang, X. Z. Wu, C. W. Wang, Z. Rong, H. M. Ding, H. Li, S. H. Li, N. S. Shao, P. T. Dong, R. Xiao and S. Q. Wang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 19958–19967.
- 48 S. Dutta Roy, M. Ghosh and J. Chowdhury, *J. Raman Spectrosc.*, 2015, **46**, 451–461.

