



Cite this: *RSC Adv.*, 2017, 7, 45478

Simple and low-cost thiophene and benzene-conjugated triarylamines as hole-transporting materials for perovskite solar cells†

Jianchang Wu,[‡] Chang Liu,[‡] Xiang Deng,^b Luozheng Zhang,^b Manman Hu,^b Jun Tang,^{ab} Wenchang Tan,^{*,a} Yanqing Tian^{‡*b} and Baomin Xu^{‡*b}

Two novel electron-rich linear small-molecules, containing benzene and thiophene as the cores with arylamine side groups, named **HTM1** and **HTM2**, respectively, were synthesized *via* short, easy and efficient synthetic routes. The influence of the π -linkers of the two materials on photophysical, electrochemical, and thermal properties, and hole mobility and photovoltaic performance was investigated. The compound with thiophene as π -linker (**HTM2**) shows better solubility and higher hole-transporting mobility than the compound with benzene as π -linker (**HTM1**). When these two materials were incorporated into perovskite solar cells as hole transporting materials (HTMs), short circuit photocurrent densities (J_{sc} s) of 15.83 mA cm⁻² and 21.1 mA cm⁻², open circuit voltages (V_{oc} s) of 0.79 V and 1.09 V, and fill factors (FFs) of 0.46 and 0.62, were obtained. These factors contributed to average overall power conversion efficiencies (PCEs) of 6.4% and 13.9% with the best PCEs of 7.5% and 14.7%, respectively. The performance of **HTM2** is comparable to the PCE obtained using the current state-of-the-art HTM of 2,2',7,7'-tetrakis(*N,N'*-di-*p*-methoxyphenylamine)-9,9'-spirobifluorene (**spiro-OMeTAD**) with the best PCE of 17.4% using a similar device preparation method and measurement conditions. These results showed that selecting a suitable π -linker is important for the performance of HTMs. And the simple **HTM2** material is a promising HTM with the potential to replace the expensive **spiro-OMeTAD** due to its comparable performance with a much simpler synthesis route and much reduced cost (10 times less than that of **spiro-OMeTAD**). This study demonstrates that a compound with a suitable π -linker could be a low-cost and high performance HTM to replace **spiro-OMeTAD**.

Received 21st July 2017
 Accepted 20th September 2017

DOI: 10.1039/c7ra08052b

rsc.li/rsc-advances

Introduction

In recent years, the efficiency of organic–inorganic metal halide perovskite-based solar cells (PSCs) has improved rapidly, and is already comparable with current commercial technologies.^{1,2} Those perovskite materials, typically methylammonium lead iodides (CH₃NH₃PbI₃), have excellent properties, including their absorption throughout the UV-vis-NIR spectrum, high extinction coefficients, large exciton diffusion lengths and ambipolar mobility. These characteristics make them excellent light harvesting materials.^{3–6} Their outstanding performance by showing an increase of the power conversion efficiency (PCE) from 3% to a certified 22% has been demonstrated during the

past few years.^{1–3,7–9} A typical PSC device comprises a TiO₂ anode, a perovskite absorbing layer, a hole transporting layer and a metallic counter electrode. In addition to perovskite absorbing materials, charge transporting layers also play an important role in charge extraction and interface modification, thus improving the performance of PSCs device.^{10,11} To date, 2,2',7,7'-tetrakis(*N,N'*-di-*p*-methoxyphenylamine)-9,9'-spirobifluorene (**spiro-OMeTAD**) is the most extensively studied and applied HTM for perovskite devices.^{12–16} Many device characterization and optimization works have been centered on **spiro-OMeTAD**.^{9,15,17} However, in generally, the complex synthetic route and purification of **spiro-OMeTAD** render it expensive, which significantly limits potential large-scale application and commercialization of PSC devices in the future. The synthesis of the core of **spiro-OMeTAD** needs four steps, including a Suzuki–Miyaura cross-coupling reaction, a Grignard reaction, a cyclization reaction and a bromination reaction from commercially available raw materials, which limited its total yield (<30%) and cost (about 94 \$ per g).^{18,19} Thus, significant research effort has been spent on designing and synthesizing alternative HTMs, which are more economical and are able to show device performance comparable to or even better than that of

^aSchool of Advanced Materials, Peking University Shenzhen Graduate School, Shenzhen, Guangdong 518055, P. R. China. E-mail: tanwch@pku.edu.cn

^bDepartment of Materials Science and Engineering, Southern University of Science and Technology, No. 1088, Xueyuan Rd, Shenzhen, Guangdong 518055, P. R. China. E-mail: tianyq@sustc.edu.cn; xubm@sustc.edu.cn

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra08052b

‡ These authors contributed equally.



spiro-OMeTAD. Several conjugated polymers including poly(3-hexylthiophene-2,5-diyl) (P3HT)²⁰ and poly[bis(4-phenyl)(2,4,6-trimethylphenyl)amine] (PTAA)⁸ have been employed as HTMs in perovskite solar cells. However their device results to date have been inferior to that of **spiro-OMeTAD**. What's more, the intrinsic batch to batch variation issues of polymers due to different polymerization degrees and polydispersity could hamper the performance reproducibility for possible commercialization of the device. On the other hand, a number of small molecule HTMs have also been reported, including spirobifluorenes derivatives,^{19,21–23} thiophene derivatives,^{24–26} triphenylamines,^{27,28} and triazatruxenes.^{29–31}

Most of them were namely decorated with diarylamines, triarylamines and/or carbazole derivatives. The best performance of those HTMs show similar or superior performance to **spiro-OMeTAD** in some aspects but no material has yet emerged clearly superior in all aspects of device performance, particularly in terms of cost, device efficiency and stability. Hence it remains a great challenge to develop new, easily synthesized, low cost and high performance HTMs for PSC devices.

An efficient way to simplify the synthesis of the HTMs is to change the spiro-core of the **spiro-OMeTAD** to a planar unit, such as diphenyl or terphenyl. In this way, only one or two synthetic steps are needed from the commercially available raw materials, this could reduce its cost significantly. In this article, 2,5-bis(4,4'-bis(methoxyphenyl)aminophen-4''-yl)-benzene (**HTM1**, its structure is given in Fig. 1) was designed and synthesized using terphenyl to replace the spiro-core of the **spiro-OMeTAD**. It only needs two synthetic steps to get the final product. However, when the material changes from spiro-structure to a planar structure, a great decrease in solubility and slight decrease in mobility occur. Those factors affect the process of the HTM film, as a result the PCE of the devices based on **HTM1** were much lower than that of **spiro-OMeTAD**. To solve the low mobility of **HTM1**, 2,5-bis(4,4'-bis(methoxyphenyl)aminophen-4''-yl)-thiophene (**HTM2**) was designed and synthesized using thiophene unit to replace the center unit (benzene) of the **HTM1**. Thiophene moiety has been widely used in organic electronics because of its excellent properties due to its high charge mobility. Unfortunately, during the manuscript preparation, a recent paper reported the use of **HTM2** as hole transporting layer in PSCs device³² with an extremely low PCE (4.31%). For this performance, it seems that **HTM2** is far from

being useful for high efficient devices and a candidate to replace **spiro-OMeTAD**. However, in our studies, we achieved much higher PCE of 14.7%. This result may be caused by the different components of perovskite layer. FAMAPbBr₃ was used in this article and MAPbI₃ was used in the ref. 32. It is obvious that the device based on FAMAPbBr₃ could achieve higher performance. (Devices based on **spiro-OMeTAD** achieve an efficiency of 17.4% in this article; it was only 11.9% in the reference). Moreover, FAMAPbBr₃ has a deeper HOMO level compared with MAPbI₃, which could match **HTM2** better. Further, we compared the cost of the synthesis of **HTM2** with **spiro-OMeTAD**. Because of the much shorter synthesis route of **HTM2** than that of **spiro-OMeTAD**, the cost of **HTM2** is about 1/10 fold of **spiro-OMeTAD**. Thus, our results demonstrated that the **HTM2** has a great potential to replace the **spiro-OMeTAD**.

Results and discussions

The synthetic route (Scheme S1†) towards **HTM1** and **HTM2** only requires two steps and uses commercially available raw materials, which are much simpler than that of the synthesis of **spiro-OMeTAD**. Pd-catalyzed boronation of 4-bromo-*N,N*-bis(4-methoxyphenyl)aniline (compound 1, ESI Scheme S1†) with bis(pinacolato)diboron produced the key building block OMT BPIN (compound 2, Scheme S1†) in a yield of 80%. The two final HTMs were synthesized by Suzuki coupling reaction using OMT BPIN with dibromobenzene (compound 3, Scheme S1†) and dibromothiophene (compound 4, Scheme S1†), respectively. **HTM2** has a good solubility in common solvents, such as chlorobenzene (200 mg mL⁻¹); while the solubility of **HTM1** is relatively low, about 60 mg mL⁻¹ in chlorobenzene. The good solubility of **HTM2** will benefit the smooth film formation. Differential scanning calorimetry (DSC) measurements show that **HTM1** and **HTM2** have a glass transition temperature (*T*_g) at 80 °C and 76 °C, respectively. Both are much lower than that of **spiro-OMeTAD** (around 125 °C).³³ The lower *T*_g and higher solubility could increase the pore-filling of the HTM in mesoscopic architecture device, because the pore filling procedure begins by spin-coating a solution of the HTM onto the *m*-TiO₂/perovskite film.³³ As solvent evaporates and the solution concentration increases, HTM diffuses into the pores allowing for a higher degree of pore filling than that would be obtained from the initial concentration of the solution alone. When the concentration increased to a certain degree beyond the solubility of HTM, the diffusion stops, thus higher solubility contributes to better pore-filling of the device. And a lower *T*_g could help to increase the pore-filling through a melt infiltration process. The device with a high pore-filling may have a high coverage of perovskite with HTM, thus yields a higher charge injection efficiency and lower electron-hole recombination.³³

The absorption spectra (Fig. 2a and Table 1) indicate that **HTM1** starts to absorb light (λ_{onset}) from λ at 405 nm, which is 8 nm blueshifted comparing to that of **spiro-OMeTAD**. While the λ_{onset} value of **HTM2** is 451 nm, which is 38 nm redshifted from that of **spiro-OMeTAD** (46 nm redshifted from that of **HTM1**). This is attributed to the higher electron density of the backbone for **HTM2** compared with **HTM1**.

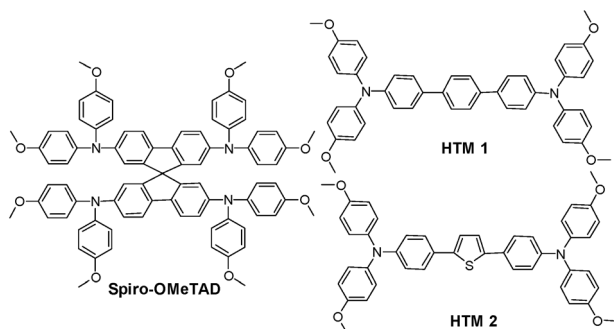


Fig. 1 Chemical structures of **spiro-OMeTAD**, **HTM1** and **HTM2**.



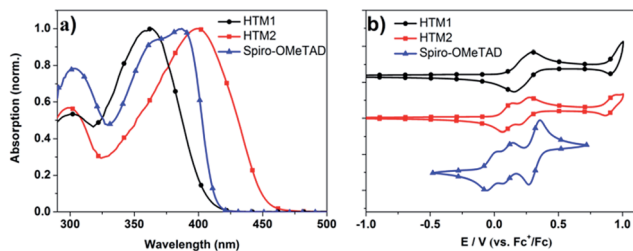


Fig. 2 (a) Normalized UV-vis absorption spectra of HTM1, HTM2 and spiro-OMeTAD in THF solution; (b) cyclic voltammograms of HTM1, HTM2 and spiro-OMeTAD in dry dichloromethane-TBAPF₆ (0.1 M); scan speed: 100 mV s⁻¹, potentials vs. Fc/Fc⁺.

Table 1 Spectroscopic and electrochemical data of HTM1, HTM2 and spiro-OMeTAD

HTMs	T _g [°C]	λ _{onset} [nm]	E _g ^a [eV]	E _{HOMO} ^b [eV]	E _{LUMO} ^c [eV]
HTM1	80	405	3.06	-5.21	-2.15
HTM2	76	451	2.75	-5.13	-2.38
Spiro-OMeTAD	125	413	3.0	-5.01	-2.01

^a Estimated from the absorption onset wavelength, E_g (eV) = 1240/λ_{onset} (nm). ^b Calculated from $E_{HOMO} = -(E_{onset}^{ox} + 5.10)$ (eV).³⁴ ^c E_{LUMO} calculated by $E_{LUMO} = E_{HOMO} + E_g$.

Cyclic voltammetric measurements were carried out to analyze and compare the energy levels of the two new HTMs and spiro-OMeTAD for the favourable hole injection process from the perovskite layer. Fig. 2b depicts the cyclic voltammograms (CV) in dichloromethane solution (1.0×10^{-3} mol L⁻¹) with tetrabutylammonium hexafluorophosphate (TBAPF₆, 0.1 mol L⁻¹) at a scan rate of 100 mV s⁻¹. As can be seen from this figure, the HOMO levels of HTM1 and HTM2 are -5.21 and -5.13 eV, respectively. Both are 0.10–0.20 eV deeper than that of spiro-OMeTAD (-5.01 eV). The HOMO level of HTM1 is 0.08 eV deeper than that of HTM2, which is attributed to the higher electron density of the backbone for HTM2. The deeper HOMO of the new HTMs could benefit the higher open-circuit voltage (V_{oc}) of the device. In principle, V_{oc} of a conventional solid-state dye sensitized solar cell (DSSC) is determined by the difference between the quasi-Fermi levels of the electrons in TiO₂ and the holes in the HTM.³⁵ And this is the same for PSCs devices. So from the electrochemical experiments data, we can expect that the device using those new HTMs would have a higher V_{oc} than that of spiro-OMeTAD.

To investigate the charge-carrier mobility of HTM1, HTM2 and spiro-OMeTAD, hole-only devices were fabricated with device structure of fluorine-doped tin oxide (FTO) glasses/(poly(3,4-ethylenedioxythiophene):polystyrene, PEDOT:PSS)/doped HTM1 or doped HTM2 (or doped spiro-OMeTAD)/Au. Hole mobilities of HTM1 and HTM2 were measured *via* space charge limitation of current (SCLC) under dark conditions, as showed in Fig. S9.† The results calculated from the Mott-Gurney law shows that by replacing benzene unit with thiophene unit, the mobility of the material increased about three times (3.1×10^{-5} cm² V⁻¹ S⁻¹ for

HTM1 and 9.0×10^{-5} cm² V⁻¹ S⁻¹ for HTM2). However, HTM1 and HTM2 both have a lower mobility than spiro-OMeTAD (3.7×10^{-4} cm² V⁻¹ S⁻¹), which suggest that HTM1 and HTM2 based devices should show slightly lower fill factors.

The new semiconductors HTM1 and HTM2 were tested as HTMs in perovskite solar cells. The device structure can be seen from the scanning electron microscope (SEM) image of the device cross-section (Fig. 3a and b). At the anode, a compact and mesoporous layer of TiO₂ was deposited on top of a fluorine-doped tin oxide (FTO) coated glass and used as the electron collector, while the cathode consists of a gold electrode thermally evaporated onto the HTM layer. The photoactive material FAPAPbBrI₃ is sandwiched between the anode and the cathode. The current-voltage (J - V) characteristics of perovskite solar cells that employ HTM1, HTM2, and spiro-OMeTAD, respectively, as the HTMs are shown in Fig. 3d and summarized in Table 2. The PCEs of the best performing devices with HTM1, HTM2, and spiro-OMeTAD as HTMs are 7.5%, 14.7% and 17.4%, respectively. Correspondingly, the short-circuit current densities (J_{sc}) of the cells are 15.8, 21.1 and 23.5 mA cm⁻², respectively. It is clear that J_{sc} value of HTM2 is as high as that spiro-OMeTAD, while J_{sc} of HTM1 is relatively low. The open-circuit voltage (V_{oc}) of HTM2 (1.09 V) is also higher than that of spiro-OMeTAD. The only different parameters are the fill factors (FFs). For the device based on spiro-OMeTAD the FF could achieve 0.72, while FF is only 0.62 for HTM2, which contributes to the slightly low PCEs for HTM2. HTM1 has a deeper HOMO energy level, but the V_{oc} has not been improved, only 0.79 V. And its FF is also relatively low, 0.46. Those may be caused by its poor mobility, limiting the charge transport in the film.

To investigate the charge transfer at the interface between the perovskite and the hole-transport materials, steady-state and time-resolved photoluminescence (PL) spectra were recorded and these are shown in Fig. 4. From the steady-state PL spectra, the perovskite film without HTMs shows a strong PL signal due to the recombination between the photo-generated electrons and holes. After deposition of HTMs, a dramatic decrease of PL intensity is observed, especially for the HTM2

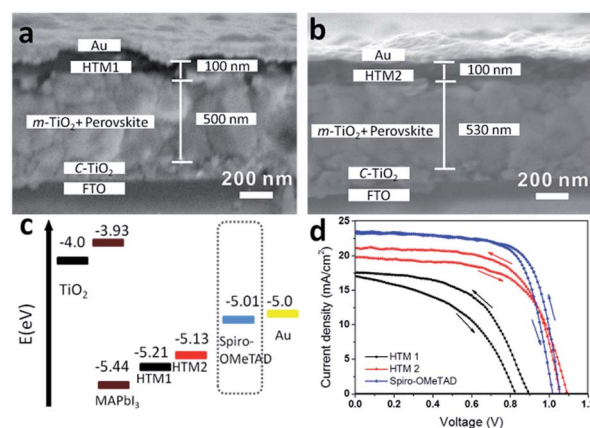


Fig. 3 Cross-sectional SEM pictures of PSCs devices based on HTM1 (a) and HTM2 (b); and the relative energy levels in the PSCs (c) and current-voltage curves (d) of PSCs based on different HTMs.



Table 2 Summary of device parameters; open-circuit voltage (V_{oc}), short-circuit current density (J_{sc}), fill factor (FF) and average efficiency (PCE)

HTM	J_{sc} [mA cm ⁻²]	V_{oc} [V]	FF	PCE [%]
HTM1	15.83 ± 1.46	0.79 ± 0.06	0.46 ± 0.04	6.33 ± 1.21%
HTM2	21.13 ± 0.85	1.05 ± 0.04	0.61 ± 0.02	13.93 ± 0.77%
Spiro-MeOTAD	22.22 ± 0.53	1.04 ± 0.02	0.69 ± 0.05	17.02 ± 0.38%

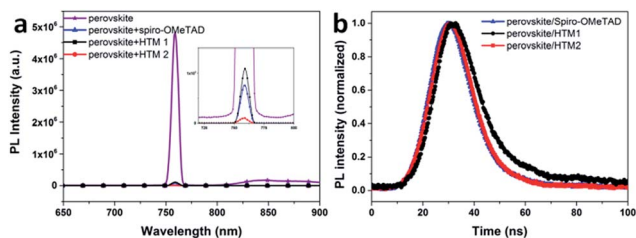


Fig. 4 (a) Photoluminescence (PL) spectra of perovskite film, perovskite/spiro-OMeTAD film, perovskite/HTM1 film, perovskite/HTM2 film; (b) time-resolved PL spectra of perovskite/spiro-OMeTAD film, perovskite/HTM1 film, perovskite/HTM2 film.

based devices. A more efficient quenching of the photo-generated charges can lead to faster interface charge separation in **HTM2** based devices than that of **HTM1** based devices, contributing to a better J_{sc} . However, the lower SCLC hole mobility limited its charge transporting compared to **spiro-OMeTAD**. Time-resolved PL measurements were carried out with excitation at 530 nm and monitoring the entire emission spectral range (Fig. 4b). Perovskite/**spiro-OMeTAD** samples showed a decay time of 3.5 ns, whereas perovskite/**HTM1** and perovskite/**HTM2** samples showed a longer decay lifetime of 4.3 ns and 4.2 ns, respectively. This indicates that **HTM2** has slightly better hole extraction ability as compared to **HTM1**. The lower interface charge separation and poorer hole extraction ability of **HTM1** could explain the low performance of the device based on **HTM1** to some extent.

In order to compare the cost of the synthesized compound **HTM1** and **HTM2** with that of **spiro-OMeTAD**, estimations were made using a method published by Docampo *et al.*³⁶ (see the Table S1†). As shown in Table 3, the cost of materials (including reagents, solvents and other consumables) required for the

Table 3 Comparison of the estimated material costs

HTM	Material cost ^a (\$ per g)	Commercial price (\$ per g)
HTM1	9.3	—
HTM2	8.6	—
Spiro-MeOTAD	93.6 ^b	173.5–484.9

^a Estimated material cost for the synthesis of 1 g of product. ^b Taken from ref. 36. The cost unit was in dollar (\$).

synthesis of 1 g of **HTM1** and **HTM2** are 9.3 \$ and 8.6 \$, respectively, which are more than 10 times lower than that of **spiro-OMeTAD**. The main reason lies in the low cost starting materials and the simpler synthesis route. Hence it has a great potential to be commercialized for **HTM2**.

Conclusions

In conclusion, two simple triphenylamine-based HTMs were developed by varying the core structure. It is found that thiophene unit played an important role not only in increasing solubility, but also in enhancing hole mobility and charge extraction, thus improving the PSCs performance. The PSCs device based on HTM with thiophene (**HTM2**) achieved higher V_{oc} , J_{sc} and FF values than that of HTM with benzene (**HTM1**). These results showed that selecting a proper π -linker is important for the performance of the HTMs. In addition, both the two HTMs could be synthesized simply with high yields by using two steps from commercially available starting materials, making those HTMs very promising for large-scale industrial production.

Experimental section

Materials

All reagents and solvents, unless otherwise specified, were purchased from Aladdin, Alfa, Aldrich and J&K Scientific Ltd. and were used without further purification. 4-Bromo-*N,N'*-bis(4-methoxyphenyl)aniline was purchased from Sdyano fine chemical Co. Ltd. mesoporous TiO₂ paste (Dyesol, 30NR-D), CsI, FAI, PbI₂, MABr and PbBr₂ were purchased from Xi'an Polymer Light Technology Corp., and 2,2',7,7'-tetrakis(*N,N*-di-*p*-methoxyphenylamine)-9,9'-spirobifluorene (**spiro-OMeTAD**) was purchased from Feiming Chemical Limited.

Measurement and characterization

Routine nuclear magnetic resonance (NMR) spectra were recorded on a Bruker Avance III 400 MHz (400 and 100 MHz for ¹H and ¹³C NMR, respectively). Chemical shifts were reported as δ values (ppm) with tetramethylsilane (TMS) as the internal standard. The splitting patterns are designated as follows: singlet (s), doublet (d), triplet (t), and multiplet (m). UV-vis absorption spectra were recorded on a PerkinElmer Lambda650 spectrophotometer from dilute solutions in THF. Cyclic voltammetry (CV) was obtained in a tetrabutylammonium hexafluorophosphate (TBAPF₆, 0.1 mol L⁻¹) supported CH₂Cl₂ solution at room temperature using a CHI600E electrochemical workstation (CH Instrument Ins.) operated at a scanning rate of 100 mV s⁻¹. A Pt wire ($\phi = 1.0$ mm) embedded in Teflon column was used as the working electrode, and a Pt sheet and Ag/AgCl electrodes were served as the counter and reference electrodes, respectively. Ferrocene/ferrocenium was used as the internal reference to calibrate the redox potentials. Thermo-gravimetric analysis (TGA) and differential scanning calorimetry (DSC) were carried out with a Mettler-Toledo TGA STAR^c/DSC1 thermal analyzer under purified nitrogen gas flow



with a 10 °C min⁻¹ heating rate. For time resolved fluorescence spectroscopy, a time-correlated single photon counting (TCSPC) system was used. The samples were excited with a picosecond diode laser (Edinburgh Instruments, EPL900) at 530 nm. Photocurrent density–voltage (*J*–*V*) curves were measured under AM 1.5G one sun illumination (100 mW cm⁻²) with a solar simulator (Enlitech SS-F7-3A) equipped with 300 W xenon lamp and a Keithley 2400 source meter. The light intensity was adjusted by an NREL-calibrated Si solar cell. The device active area was 0.08 cm⁻². The *J*–*V* curves were recorded from –0.2 V to 1.2 V with 0.01 V steps, integrating the signal for 10 ms after a 10 ms delay. The average *V*_{oc}, *J*_{sc} and FF were based on 10 devices.

Device fabrication

Fluorine-doped tin oxide (FTO) glasses (Nippon Sheet Glass) were cleaned with detergent, deionized water, and acetone under sonication with ethanol in an ultrasonic bath for 30 minutes. After that, FTO glasses were treated in a UV cleaner for 30 minutes. The cleaned FTO glasses were coated with 0.15 M titanium diisopropoxide bis(acetylacetonate) in 1-butanol solution by the spin-coating method at 2000 rpm for 60 s, followed by heating at 125 °C for 5 minutes. The films were cooled down to room temperature and a 0.15 M titanium diisopropoxide bis(acetylacetonate) solution in 1-butanol was spin coated again to make a pin hole free dense TiO₂ film. And then the substrates were calcined in a box furnace at 450 °C for 30 minutes. Mesoporous TiO₂ (mp-TiO₂) layer was deposited on the dense TiO₂ layer by spin-coating the TiO₂ ethanol solution containing 14.3 wt% TiO₂ paste at 4000 rpm for 20 seconds, which was then calcined at 500 °C for 0.5 hour. CsI (0.05 M), FAI (1 M), PbI₂ (1.1 M), MABr (0.2 M) and PbBr₂ (0.2 M) was dissolved in anhydrous DMF:DMSO 9 : 1 (v/v), which was heated at 70 °C for 2 hours under magnetic stirring, and the mixture was then spin coated on the as-prepared TiO₂ film at 4000 rpm for 40 seconds a nitrogen atmosphere glove box. During the spin coating step, 100 μL of chlorobenzene was poured on the spinning substrate 15 s prior to the end of the program. Afterward, the as-prepared films were heated at 100 °C for around 60 min until the color changed to dark red. The hole-transporting layer (HTL) was prepared by spin-coating an HTM solution at 4000 rpm for 20 s. The HTM solution was prepared by dissolving **spiro-OMeTAD**, **HTM1** and **HTM2** in chlorobenzene (60 mM) with 28.8 μL tBP solution and 17.5 μL Li-bis(trifluoromethanesulfonyl) imide (Li-TFSI)/acetonitrile (520 mg/1 mL). Devices were then left overnight in air. Finally, a 100 nm thick gold (Au) layer was thermally evaporated under vacuum of 4 × 10⁻⁵ Torr, at a rate of ~0.05 nm s⁻¹, to finish the device fabrication.

Synthesis

***N,N*-Bis(4-methoxyphenyl)-4-(4,4,5,5-tetramethyl-1,3,2-dioxaborolan-2-yl)benzamine (OMT BPIN, 2, Scheme S1†)**. OMT BPIN was synthesized according to the reported literature.³⁷ A solution of 4-bromo-*N,N*-bis(4-methoxyphenyl)aniline (2.0 g, 5.2 mmol), bis(pinacolato)diboron (2.0 g, 7.9 mmol), Pd(dppf)Cl₂ (0.23 g, 0.31 mmol), and KOAc (1.5 g, 15.6 mmol) in dioxane (30 mL) was stirred at 70 °C for 20 h under N₂. The

reaction process was monitored by TLC. It was found that the starting material was consumed after 20 h. After cooled to room temperature, the reaction was poured in to CH₂Cl₂ (100 mL), extracted with CH₂Cl₂, washed with water. The crude product was purified by recrystallization with methanol to give product (1.8 g, 80%) as yellow solid. ¹H NMR (400 MHz, CDCl₃, δ): 7.60 (br, 2H), 7.05 (br, 4H), 6.83 (d, *J* = 8.0 Hz, 6H), 3.80 (s, 6H), 1.32 (s, 12H).

2,5-Bis(4,4'-bis(methoxyphenyl)aminophen-4''-yl)-benzene (HTM1). OMT BPIN (0.97 g, 2.2 mmol), 1,4-dibromobenzene (0.24 g, 1.0 mmol), Pd(PPh₃)₄ (0.13 g, 0.12 mmol) was mixed in degassed THF (10 mL), degassed K₂CO₃ (2.0 mol L⁻¹ in H₂O, 2.6 mL) aqueous solution was added at the room temperature. Then the reaction mixture was stirred at reflux for 20 h. The reaction mixture was cooled to RT and poured into CH₂Cl₂, extracted with CH₂Cl₂, washed with water. The CH₂Cl₂ layer was dried over sodium sulfate and filtered. The organic solvent was removed by rotary evaporation, and the residue was purified by column chromatography (silica gel, petroleum ether/CH₂Cl₂ = 3/2, *R*_f = 0.3) to give product as yellow solid (0.51 g, 75%). ¹H NMR (400 MHz, CD₂Cl₂) δ = 7.63 (s, 4H), 7.50 (d, *J* = 8.9 Hz, 4H), 7.12 (d, *J* = 8.8 Hz, 8H), 7.00 (d, *J* = 8.5 Hz, 4H), 6.89 (d, *J* = 8.8 Hz, 8H), 3.83 (s, 12H). ¹³C NMR (100 MHz, CD₂Cl₂) δ = 156.1, 148.2, 140.8, 138.8, 132.3, 127.1, 126.7, 126.5, 120.6, 114.7, 55.4. MS (HRESI): *m/z* calcd for C₄₆H₄₀N₂O₄: 684.2983; found: 684.2985.

2,5-Bis(4,4'-bis(methoxyphenyl)aminophen-4''-yl)-thiophene (HTM2). OMT BPIN (0.97 g, 2.2 mmol), 2,5-dibromothiophene (0.25 g, 1.0 mmol), Pd(PPh₃)₄ (0.06 g, 0.05 mmol) was mixed in degassed THF (10 mL), degassed K₂CO₃ (2.0 mol L⁻¹ in H₂O, 2.6 mL) aqueous solution was added at the room temperature. Then the reaction mixture was stirred at reflux for 12 h. The reaction mixture was cooled to RT and poured into CH₂Cl₂, extracted with CH₂Cl₂, and then washed with water. The CH₂Cl₂ layer was dried over sodium sulfate and filtered. The organic solvent was removed by rotary evaporation, and the residue was purified by column chromatography (silica gel, petroleum ether/CH₂Cl₂ = 3/2, *R*_f = 0.3) to give product as yellow powder (0.59 g, 85%). ¹H NMR (400 MHz, DMSO-*d*₆): δ = 7.45 (d, *J* = 8.6 Hz, 4H), 7.28 (s, 2H), 7.04 (d, *J* = 8.8 Hz, 8H), 6.92 (d, *J* = 8.9 Hz, 8H), 6.77 (d, *J* = 9.4 Hz, 4H), 3.74 (s, 13H). ¹³C NMR (100 MHz, DMSO-*d*₆): δ = 156.3, 148.2, 141.9, 140.3, 127.2, 126.4, 126.0, 123.5, 119.9, 115.4, 55.7. MS (HRESI): *m/z* calcd for C₄₄H₃₈N₂O₄S: 690.2547; found: 690.2549.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Key Research and Development Project funding from the Ministry of Science and Technology of China (Grants No. 2016YFA0202400 and 2016YFA0202404), and the Peacock Team Project funding from Shenzhen Science and Technology Innovation Committee (Grant No. KQTD2015033110182370).



References

- J. Burschka, N. Pellet, S. J. Moon, R. Humphry-Baker, P. Gao, M. K. Nazeeruddin and M. Gratzel, *Nature*, 2013, **499**, 316–319.
- N. Research Cell Efficiency Records, <http://www.nrel.gov/ncpv/>, accessed: March 2016.
- A. Kojima, K. Teshima, Y. Shirai and T. Miyasaka, *J. Am. Chem. Soc.*, 2009, **131**, 6050–6051.
- S. D. Stranks, G. E. Eperon, G. Grancini, C. Menelaou, M. J. Alcocer, T. Leijtens, L. M. Herz, A. Petrozza and H. J. Snaith, *Science*, 2013, **342**, 341–344.
- G. Xing, N. Mathews, S. Sun, S. S. Lim, Y. M. Lam, M. Gratzel, S. Mhaisalkar and T. C. Sum, *Science*, 2013, **342**, 344–347.
- T. Salim, S. Sun, Y. Abe, A. Krishna, A. C. Grimsdale and Y. M. Lam, *J. Mater. Chem. A*, 2015, **3**, 8943–8969.
- H. Zhou, Q. Chen, G. Li, S. Luo, T. B. Song, H. S. Duan, Z. Hong, J. You, Y. Liu and Y. Yang, *Science*, 2014, **345**, 542–546.
- W. S. Yang, J. H. Noh, N. J. Jeon, Y. C. Kim, S. Ryu, J. Seo and S. I. Seok, *Science*, 2015, **348**, 1234–1237.
- M. Saliba, T. Matsui, J.-Y. Seo, K. Domanski, J.-P. Correa-Baena, M. K. Nazeeruddin, S. M. Zakeeruddin, W. Tress, A. Abate, A. Hagfeldt and M. Gratzel, *Energy Environ. Sci.*, 2016, **9**, 1989–1997.
- L. Calio, S. Kazim, M. Gratzel and S. Ahmad, *Angew. Chem., Int. Ed.*, 2016, **55**, 14522–14545.
- M. Salado, J. Idigoras, L. Calio, S. Kazim, M. K. Nazeeruddin, J. A. Anta and S. Ahmad, *ACS Appl. Mater. Interfaces*, 2016, **8**, 34414–34421.
- U. B. Cappel, E. A. Gibson, A. Hagfeldt and G. Boschloo, *J. Phys. Chem. C*, 2009, **113**, 6275–6281.
- I. K. Ding, N. Tétreault, J. Brillet, B. E. Hardin, E. H. Smith, S. J. Rosenthal, F. Sauvage, M. Grätzel and M. D. McGehee, *Adv. Funct. Mater.*, 2009, **19**, 2431–2436.
- N. J. Jeon, H. G. Lee, Y. C. Kim, J. Seo, J. H. Noh, J. Lee and S. I. Seok, *J. Am. Chem. Soc.*, 2014, **136**, 7837–7840.
- D. Liu and T. L. Kelly, *Nat. Photonics*, 2014, **8**, 133–138.
- W. H. Nguyen, C. D. Bailie, E. L. Unger and M. D. McGehee, *J. Am. Chem. Soc.*, 2014, **136**, 10996–11001.
- F. Hao, C. C. Stoumpos, D. H. Cao, R. P. H. Chang and M. G. Kanatzidis, *Nat. Photonics*, 2014, **8**, 489–494.
- T. P. Saragi, T. Spehr, A. Siebert, T. Fuhrmann-Lieker and J. Salbeck, *Chem. Rev.*, 2007, **107**, 1011–1065.
- B. Xu, D. Bi, Y. Hua, P. Liu, M. Cheng, M. Grätzel, L. Kloo, A. Hagfeldt and L. Sun, *Energy Environ. Sci.*, 2016, **9**, 873–877.
- Y. Guo, C. Liu, K. Inoue, K. Harano, H. Tanaka and E. Nakamura, *J. Mater. Chem. A*, 2014, **2**, 13827–13830.
- D. Q. Bi, B. Xu, P. Gao, L. C. Sun, M. Graetzel and A. Hagfeldt, *Nano Energy*, 2016, **23**, 138–144.
- M.-H. Li, C.-W. Hsu, P.-S. Shen, H.-M. Cheng, Y. Chi, P. Chen and T.-F. Guo, *Chem. Commun.*, 2015, **51**, 15518–15521.
- M. Saliba, S. Orlandi, T. Matsui, S. Aghazada, M. Cavazzini, J.-P. Correa-Baena, P. Gao, R. Scopelliti, E. Mosconi, K.-H. Dahmen, F. De Angelis, A. Abate, A. Hagfeldt, G. Pozzi, M. Graetzel and M. K. Nazeeruddin, *Nat. Energy*, 2016, **1**, 15017.
- H. Li, K. Fu, A. Hagfeldt, M. Grätzel, S. G. Mhaisalkar and A. C. Grimsdale, *Angew. Chem., Int. Ed.*, 2014, **53**, 4085–4088.
- T. Krishnamoorthy, F. Kunwu, P. P. Boix, H. Li, T. M. Koh, W. L. Leong, S. Powar, A. Grimsdale, M. Grätzel, N. Mathews and S. G. Mhaisalkar, *J. Mater. Chem. A*, 2014, **2**, 6305.
- A. Krishna, D. Sabba, H. Li, J. Yin, P. P. Boix, C. Soci, S. G. Mhaisalkar and A. C. Grimsdale, *Chem. Sci.*, 2014, **5**, 2702–2709.
- H. Choi, S. Park, S. Paek, P. Ekanayake, M. K. Nazeeruddin and J. Ko, *J. Mater. Chem. A*, 2014, **2**, 19136–19140.
- H. Choi, S. Paek, N. Lim, Y. H. Lee, M. K. Nazeeruddin and J. Ko, *Chem. – Eur. J.*, 2014, **20**, 10894–10899.
- F. J. Ramos, K. Rakstys, S. Kazim, M. Gratzel, M. K. Nazeeruddin and S. Ahmad, *RSC Adv.*, 2015, **5**, 53426–53432.
- K. Rakstys, A. Abate, M. I. Dar, P. Gao, V. Jankauskas, G. Jacopin, E. Kamarauskas, S. Kazim, S. Ahmad, M. Grätzel and M. K. Nazeeruddin, *J. Am. Chem. Soc.*, 2015, **137**, 16172–16178.
- L. Calió, C. Momblona, L. Gil-Escrig, S. Kazim, M. Sessolo, Á. Sastre-Santos, H. J. Bolink and S. Ahmad, *Sol. Energy Mater. Sol. Cells*, 2017, **163**, 237–241.
- X. Liu, F. Kong, F. Guo, T. Cheng, W. Chen, T. Yu, J. Chen, Z. A. Tan and S. Dai, *Dyes Pigm.*, 2017, **139**, 129–135.
- T. Leijtens, I. K. Ding, T. Giovenzana, J. T. Bloking, M. D. McGehee and A. Sellinger, *ACS Nano*, 2012, **6**, 1455–1462.
- Y. Hua, B. Xu, P. Liu, H. Chen, H. Tian, M. Cheng, L. Kloo and L. Sun, *Chem. Sci.*, 2016, **7**, 2633–2638.
- A. Hagfeldt, G. Boschloo, L. Sun, L. Kloo and H. Pettersson, *Chem. Rev.*, 2010, **110**, 6595–6663.
- M. L. Petrus, T. Bein, T. J. Dingemans and P. Docampo, *J. Mater. Chem. A*, 2015, **3**, 12159–12162.
- L. Chen, B. H. Zhang, Y. X. Cheng, Z. Y. Xie, L. X. Wang, X. B. Jing and F. S. Wang, *Adv. Funct. Mater.*, 2010, **20**, 3143–3153.

