RSC Advances

PAPER

Cite this: RSC Adv., 2017, 7, 42088

Received 8th August 2017 Accepted 25th August 2017

DOI: 10.1039/c7ra08746b

rsc.li/rsc-advances

Introduction

 $Co₃O₄$ is attracting much attention for its thermodynamic stability, electrochemical properties and antiferromagnetic properties, promising for catalyst, Li-ion battery, gas sensor and spintronic device applications.¹⁻⁶ Both the experimental and theoretical studies have revealed that the crystallographic orientation and morphology of the metallic oxides play a key role in the related physical and chemical properties as different orientations or morphologies not only have various surface atomic densities but also electronic structures.⁷⁻¹¹ Hu et al.¹² found significant effects of crystal orientation in $Co₃O₄$ on catalytic properties. It was also demonstrated that the exposed facets of $Co₃O₄$ nanocrystals play an important role in Li-ion battery.¹³ Moreover, Yu et al.¹⁴ reported that the electrochemical sensing performance in (111) facet of $Co₃O₄$ nanoplates is better than that of the (001) facet of $Co₃O₄$ nanocubes.

Thereby, much effort has been made to synthesize various oriented single nanocrystals (such as nanoparticles, nanowires and nanosheets) by chemical solution deposition method, magnetron sputtering, pulsed laser deposition, post calcination, thermal evaporation, wet chemical synthetic approach, and so on.¹⁵–²¹ However, a synthesis method to produce

Orientation-controllable growth of $Co₃O₄$ single nanocrystals using a $BiCoO₃$ target by pulsed laser deposition

Deyang Chen[,](http://orcid.org/0000-0002-8370-6409) $\mathbf{D}^{\star ab}$ Shang-Lin Hsu,^b Ajay K. Yadav \mathbf{D}^{b} and James D. Clarkson^b

We report a novel route to synthesize $Co₃O₄$ single nanocrystals using pulsed laser deposition by decomposition of BiCoO₃. The decomposition of BiCoO₃ to Bi₂O₃ and Co₃O₄ at relatively high temperature enables the formation of pure $Co₃O₄$ phase. The absence of Bi element is confirmed by both X-ray diffraction (XRD) and energy dispersive spectrometer (EDS) measurements. The effects of various PLD growth conditions, including growth rate (2-20 Hz), growth temperature (630–650 °C), O_2 partial pressure (50–200 mTorr) and laser energy density (0.6–1.8 J cm⁻²), on the synthesis of Co₃O₄ nanocrystals were systematically studied. Interestingly, the orientation of $Co₃O₄$ single nanocrystals can be controlled by changing the growth parameters. It is revealed that the decrease of the laser energy density leads to the preferred orientation of $Co₃O₄$ single nanocrystals altering from (111) plane to (220), while the high temperature growth favors (400) orientation. The obtained orientation-controllable $Co₃O₄$ single nanocrystals provide the possibility to study its orientation-dependent physical and chemical properties as catalysts, anode materials and magnetic materials. PAPER
 (A) Check for updates
 **Content of the controllable growth of Co₃O₄ single

Check for updates

Content at Content and and Content and Content and Content and Content and Content and the expectation of the co**

orientation-controllable and morphology-tunable $Co₃O₄$ single nanocrystals is extremely desirable as it could provide a platform for the study of orientation-dependent and morphologyrelated catalytic, magnetic and electrochemical properties where enhanced physical or chemical properties may be obtained for promising applications.

In this study, a novel route is proposed to synthesize $Co₃O₄$ single nanocrystals using a $BiCoO₃$ target by pulsed laser deposition (PLD). We would like to mention that the term of "single nanocrystals" here presents the epitaxial grew $Co₃O₄$ nanoparticles. We show that the orientations of the $Co₃O₄$ can be well-controlled by change the PLD growth parameters (such as growth rate, growth temperature, $O₂$ partial pressure and laser energy density). The decomposition of $BiCoO₃$ to $Bi₂O₃$ and $Co₃O₄$ at relatively high temperature (630 °C) enables the formation of pure $Co₃O₄$ phase. The absence of Bi element is confirmed by both X-ray diffraction (XRD) and energy dispersive spectrometer (EDS) measurements. We systematically studied the effects of various PLD growth conditions, including growth rate (2-20 Hz), growth temperature (630–650 °C), O_2 partial pressure (50–200 mTorr) and laser energy density (0.6–1.8 J cm⁻²), on the synthesis of Co₃O₄ nanocrystals.

By changing different growth parameters, (111), (220) and (400) orientation-controlled $Co₃O₄$ single nanocrystals have been obtained. It turns out that (111) oriented $Co₃O₄$ single nanocrystals can be synthesized at slower growth rate and lower $O₂$ partial pressure, while lower growth temperature and higher $O₂$ partial pressure result in the formation of the (111) oriented $Bi₂O₃$ impurity. More interestingly, it is

YAL SOCIETY
CHEMISTRY

a Institute for Advanced Materials (IAM), South China Normal University, Guangzhou 510006, China. E-mail: dychen1987@gmail.com

b Department of Materials Science and Engineering, University of California, Berkeley, California 94720, USA

revealed that the decrease of the laser energy density leads to the preferred orientation of $Co₃O₄$ single nanocrystals altering from (111) plane to (220). Furthermore, we found that higher growth temperature would give rise to the synthesis of (400) oriented $Co₃O₄$ single nanocrystals. These findings provide a novel approach to prepare the orientationcontrollable $Co₃O₄$ single nanocrystals, which offer a possibility to study the orientation-related properties of $Co₃O₄$ for potential applications in catalysts, anode materials and magnetic materials.

Experimental

Sample preparation

A series of $Co₃O₄$ samples were grown using a BiCoO₃ target on (110) YAlO₃ substrates by pulsed laser deposition (PLD) using a KrF excimer laser ($\lambda = 248$ nm) in various growth conditions with growth rate at 2–20 Hz, growth temperature from 630 to 650 °C, O_2 partial pressure of 50–200 mTorr and laser energy density at 0.6–1.8 J cm^{-2} .

Characterizations

A combination of X-ray diffraction (XRD, PANalytical X'Pert) and transmission electron microscopy (TEM, Titan X 200 kV) were used for the structural characterization. The composition of the obtained samples were measured by energy dispersive spectrometer (EDS). Atomic force microscopy (AFM, Veeco, Multi-Mode 8) was applied to study the morphologies of $Co₃O₄$ nanocrystals.

Results and discussion

A variety of parameters play important roles in the PLD growth process. Among of these conditions, the growth rate, growth temperature, O_2 partial pressure and laser energy density are critical to obtain high quality samples.^{22,23} In this work, we first study the effects of growth rates (2–20 Hz) on the synthesis of $Co₃O₄$ single nanocrystals as shown in Fig. 1. The $Co₃O₄$ was grown by pulsed laser deposition (PLD) at various pulsed laser frequencies (2, 10 and 20 Hz) with the laser energy density of 1.1 J cm^{-2} at 630 °C in an oxygen pressure of 100 mTorr. At the growth temperature \sim 630 °C (much higher than the decomposition temperature of $BiCoO₃$), $BiCoO₃$ would decompose to $Bi₂O₃$ and $Co₃O₄$.²⁴ This is verified by the XRD measurement (Fig. 1a) and provides a possibility to obtain pure $Co₃O₄$ phase by adjusting the growth parameters. The XRD data show that (111) oriented single crystal $Co₃O₄$ with (111) $Bi₂O₃$ impurity were synthesized at low growth rate (2 Hz), while (111), (311) and (400) polycrystalline $Co₃O₄$ were obtained with the increase of the growth rate to 10 Hz. The poly-crystallization of $Co₃O₄$ were even promoted with the (220) diffraction peak emerging at 20 Hz. It is also worth to note that the amount of $Bi₂O₃$ secondary phase increases at higher growth rate (10 Hz and 20 Hz) as shown in Fig. 1a. This is because the high growth rate hinders the volatilization of Bi element, whereas the low growth rate of 2 Hz helps its volatilization to obtain pure phase $Co₃O₄$. The morphologies of the samples grown at 2 Hz, 10 Hz and 20 Hz were captured by atomic force microscopy (AFM) as shown in Fig. 1b–d, respectively. The triangle-like morphology presented in Fig. 1b points to the (111) oriented growth of Co_3O_4 nanocrystals, which is in agreement with the XRD data. This is Paper

Negro density articles. The published on $\frac{1}{2}$ and $\frac{1$

Fig. 1 The effects of growth rate on the synthesis of Co₃O₄ single nanocrystals. (a) XRD data of Co₃O₄ samples grown at various growth rate (2 Hz, 10 Hz and 20 Hz). AFM topography images showing the morphologies of the corresponding samples: (b) 2 Hz, (c) 10 Hz and (d) 20 Hz.

also consistent with previous study.²⁵ Nevertheless, the polycrystalline Co_3O_4 samples synthesized at 10 Hz and 20 Hz show particles with various shapes embedded in continuous films (Fig. 1c and d). These results indicate that pure phase, (111) oriented $Co₃O₄$ single nanocrystal favors low growth rate.

The influences of growth temperature on the preparation of $Co₃O₄$ were also studied as shown in Fig. 2. The $Co₃O₄$ samples were grown at 10 Hz with the laser energy density of 1.1 J cm⁻²

Fig. 2 Effects of growth temperate (630–650 °C) on the synthesis of $Co₃O₄$ single nanocrystals, revealing the preferred (400) oriented growth of $Co₃O₄$ in higher temperature (650 °C).

in an oxygen pressure of 100 mTorr at 630-650 $^{\circ}$ C. The XRD data demonstrate that high growth temperature contributes to the growth of (400) oriented $Co₃O₄$. However, there is the impurity $Bi₂O₃$ phase in all these samples grown at different growth rates and temperatures.

As the secondary $Bi₂O₃$ phase is sensitive to the $O₂$ partial pressure as well,^{26,27} to obtain pure $Co₃O₄$, we next focus on the effects of this growth parameter. The samples were grown at 2 Hz with 1.1 J cm⁻² energy density at 630 °C in the O_2 pressure of 50–200 mTorr. XRD was performed for the structural characterization. As expected, the O_2 pressure has significant effects on the formation of the $Bi₂O₃$ impurity. With the decrease of the O_2 pressure from 200 mTorr to 100 mTorr, the content of Bi_2O_3 reduced significantly and pure (111) oriented $Co₃O₄$ was successfully obtained with the further decrease of O_2 to 50 mTorr, as displayed in Fig. 3a. All the corresponding AFM topography images (Fig. 3b–d) show similar triangle-like morphologies in agreement with the XRD data (Fig. 3a) and the growth rate dependent study (Fig. 1), further indicating the (111) oriented growth of $Co₃O₄$ single nanocrystals. The size of the obtained $Co₃O₄$ single nanocrystals are 100–350 nm. We would like to mention that both the $Co₃O₄$ and $Bi₂O₃$ are (111)oriented single nanocrystals, thus it is difficult to tell the difference in the topography images as they both show trianglelike morphology.

The laser energy density is another important growth parameter of PLD, which would affect the formation of the desired phase and the impurities.²⁸ Interestingly, we find that this growth parameter can be used to control the orientation of $Co₃O₄$ nanocrystals. Three samples were grown with different

Fig. 3 Effects of oxygen partial pressure on the synthesis of Co₃O₄ single nanocrystals. (a) XRD data of Co₃O₄ samples grown at different O₂ pressure (50–200 mTorr). AFM topography images showing the morphologies of the corresponding samples: (b) 200 mTorr, (c) 100 mTorr and (d) 50 mTorr.

laser energy density (0.6–1.8 J $\rm cm^{-2})$ while other growth parameters were the same (50 mTorr, 2 Hz, 630 $^{\circ}$ C). As shown in Fig. 4a, it is revealed that high energy density $(1.8~\mathrm{J~cm}^{-2})$ leads to the formation of (111) and (220) mixed orientation of $Co₃O₄$ nanocrystals. (111) Single nanocrystal $Co₃O₄$ were synthesized with the reduction of energy density to 1.1 J cm $^{-2}$, whereas the preferred orientation alters to (220) plane with even lower energy density of 0.6 J $\rm cm^{-2}.$

Meanwhile, the corresponding morphologies are very different as shown in the AFM images (Fig. 4b–d). The (111) and (220) mixed orientation samples obtained at 1.8 J cm⁻² show irregular shapes (Fig. 4b) whereas the (111) oriented $Co₃O₄$ single nanocrystals grown at 1.1–1.8 J cm⁻² possess triangle-like morphology (Fig. 4c), again in consistent with the grow rate and oxygen pressure dependent studies (Fig. 1 and 3). Differently, when the orientation of $Co₃O₄$ change to (220) plane with the decrease of energy density to 0.6 J cm $^{-2}$, it shows sandwich-like morphology as shown in Fig. 4d. The typical sandwich-like $Co₃O₄$ nanocrystals are marked with blue squares while triangle-like nanocrystals marked with white circles are also observed, indicating the existence of a small amount of (111) oriented $Co₃O₄$ which was not detected by XRD. These findings are promising for the study of the morphology-dependent and orientation-dependent catalytic, magnetic and electrochemical properties.

To further study the structure and composition of the obtained $Co₃O₄$ single nanocrystals, transmission electron microscopy (TEM), Scanning TEM (STEM) and energy dispersive spectrometer (EDS) were performed on a (111) oriented $Co₃O₄$ sample as shown in Fig. 4c at the National Center for Electron

Microscopy (NCEM), Lawrence Berkeley National Laboratory (LBNL). Fig. 5a shows the high-angle annular dark field (HAADF) low-resolution STEM image of $Co₃O₄$ sample grown on $YAO₃$ substrate. The EDS elemental line scan profile is shown in Fig. 5b, which was conducted across a $Co₃O₄$ nanocrystal along the red line as shown in Fig. 5a. The corresponding EDS line scan across HAADF STEM image is presented in Fig. 5c. It reveal the existence of Co and O in the layer of $Co₃O₄$ nanocrystal and the presence of Y, Al and O in the layer of $YAIO₃$ substrate, while the Bi element from the $BiCoO₃$ target is absent due to its decomposition during the growth process. This is in agreement of our XRD data. The EDS elemental mappings of Bi, Co and O elements corresponding to the HAADF STEM image in Fig. 5a were measured as well as shown in Fig. 5d–f. It again demonstrate the absence of Bi (as shown in Fig. 5d, the discontinuous blue contrast is the background signal). Fig. 5e shows that the Co element only exists in the $Co₃O₄$ single nanocrystals layer while Fig. 5f reveals that oxygen exists across the whole sample. Thus, these results demonstrate the pure $Co₃O₄$ phase has been obtained in this study. Paper Weak control (0.6-1.8 J cm⁻²) while online growth Microscopy (NCEM), Lawrence Brehely Scalional Informery

not the formation of (111) and (29)) mixed on creation (1600). Yaki) substrate. The any change is many com

A low-resolution cross sectional TEM image of the (111) oriented $Co₃O₄$ sample (as shown in Fig. 4c) is presented in Fig. 6a. It shows that the discontinuous $Co₃O₄$ nanocrystals with size around 100–250 nm are synthesized on $YAlO₃$ substrate. The high-resolution TEM image of the red circled region in Fig. 6a is shown in Fig. 6b. The green dashed line reveals the sharp interface between the $Co₃O₄$ layer and YAlO₃ substrate. Moreover, the lattice spacing of 4.67 \AA verifies the (111) oriented growth of $Co₃O₄$ single nanocrystals, which is again in agreement with the previous data (Fig. 4).

Fig. 4 Effects of laser energy density on the synthesis of Co₃O₄ single nanocrystals. (a) XRD data of Co₃O₄ samples grown at different energy density (0.6–1.8 J cm⁻²). AFM topography images showing the morphologies of the corresponding samples: (b) 1.8 J cm⁻², (c) 1.1 J cm⁻², and (d) 0.6 J cm^{-2} .

Fig. 5 (a) Low-resolution HAADF STEM image of Co₃O₄ sample grown on YAIO₃ substrate. (b) EDS elemental line scan profile across the red line in (a). (c) The HAADF STEM image of the region conducted with EDS line scan. The EDS elemental mappings of Bi, Co and O elements corresponding to the HAADF STEM image in (a): (d) Bi element, (e) Co element, and (f) O element.

Fig. 6 (a) A low-resolution cross sectional TEM image of the (111) oriented $Co₃O₄$ sample. (b) The high-resolution TEM image of the red circled region in (a). The green dashed line reveals the sharp interface between the Co₃O₄ layer and YAIO₃ substrate. The lattice spacing of 4.67 Å verifies the (111) oriented growth of $Co₃O₄$ single nanocrystals.

Conclusions

To summarize, we report a novel approach to synthesize single nanocrystal $Co₃O₄$ using PLD by the decomposition of BiCoO₃. The absence of Bi element owing to the decomposition of $BiCoO₃$ during the growth process is verified by both X-ray diffraction (XRD) and energy dispersive spectrometer (EDS) measurements. We show that the orientation of $Co₃O₄$ can be well-controlled by changing the PLD growth conditions. (111), (220) and (400) oriented $Co₃O₄$ single nanocrystals have been

successfully obtained. It is revealed that the decrease of the laser energy density leads to the preferred orientation of Co_3O_4 single nanocrystals altering from (111) plane to (220), while the high temperature growth favors (400) orientation. Moreover, distinct morphologies of different orientation $Co₃O₄$ single nanocrystals are observed by AFM. The (111) oriented growth samples show triangle-like morphology while the (220) samples display sandwich-like topography. These findings offer a new route to obtain orientation-controllable $Co₃O₄$ single nanocrystals and provide the possibility to study the orientation-

Paper RSC Advances

dependent and morphology-dependent physical and chemical properties of $Co₃O₄$ as catalysts, anode materials and magnetic materials. Furthermore, the decomposition of $BiCoO₃$ in high temperature (630–650 °C) in this study also might give a clue to grow $BiCoO₃$ thin films, which is a superior room temperature ferroelectric material, 29 in relatively low temperature.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors would like to thank the National Science Foundation (NSF) Center for Energy Efficient Electronics Science (E³S) under grant number ECCS-0939514 and the Startup Project for Doctors of National Natural Science Foundation of Guangdong and China Postdoctoral Science Foundation (Grant No. 2017M612687). D. C. acknowledges financial support from the Oversea Study Program of Guangzhou Elite Project (GEP). Paper

Open Access Article Correll and contentring and themical 212. II. Hu, $\sqrt{9}$, May, D. D., L, *A. A.* Cheen, Sec. 2018, 120

(mapy and K. Color complete is not commonstrate the magnetic is article. Note that it is a

References

- 1 A. Alvarez, S. Ivanova, M. A. Centeno and J. A. Odriozola, Appl. Catal., A, 2012, 431, 9–17.
- 2 R. Gao, J. Z. Zhu, X. L. Xiao, Z. B. Hu, J. J. Liu and X. F. Liu, J. Phys. Chem. C, 2015, 119, 4516–4523.
- 3 K. B. Yin, J. Ji, Y. T. Shen, Y. W. Xiong, H. C. Bi, J. Sun, T. Xu, Z. Y. Zhu and L. T. Sun, J. Alloys Compd., 2017, 720, 345–351.
- 4 J. M. Xu and J. P. Cheng, J. Alloys Compd., 2016, 686, 753–768.
- 5 B. L. Ellis, P. Knauth and T. Djenizian, Adv. Mater., 2014, 26, 3368–3398.
- 6 W. Wei, Z. Wang, Z. Liu, Y. Liu, L. He, D. Chen, A. Umar, L. Guo and J. Li, J. Power Sources, 2013, 238, 376–387.
- 7 Y. Gong, F. Gong, C. Wang, H. Zheng and F. Li, RSC Adv., 2015, 5, 27266–27272.
- 8 Z. L. Wang, J. Phys. Chem. B, 2000, 104, 1153–1175.
- 9 T. Ouyang, K. Cheng, S. Kong, K. Ye, Y. Gao, D. Zhang, G. Wang and D. Cao, RSC Adv., 2015, 5, 36059–36065.
- 10 X. Xie, Y. Li, Z.-Q. Liu, M. Haruta and W. Shen, Nature, 2009, 458, 746–749.
- 11 L. Jin, X. Li, H. Ming, H. Wang, Z. Jia, Y. Fu, J. Adkins, Q. Zhou and J. Zheng, RSC Adv., 2014, 4, 6083–6089.
- 12 L. H. Hu, Q. Peng and Y. D. Li, J. Am. Chem. Soc., 2008, 130, 16136–16137.
- 13 X. Xiao, X. Liu, H. Zhao, D. Chen, F. Liu, J. Xiang, Z. Hu and Y. Li, Adv. Mater., 2012, 24, 5762–5766.
- 14 X. Y. Yu, Q. Q. Meng, T. Luo, Y. Jia, B. Sun, Q. X. Li, J. H. Liu and X. J. Huang, Sci. Rep., 2013, 3, 2886.
- 15 Z. Dong, Y. Y. Fu, Q. Han, Y. Y. Xu and H. Zhang, J. Phys. Chem. C, 2007, 111, 18475–18478.
- 16 L. F. Chen, J. C. Hu, R. Richards, S. Prikhodko and S. Kodambaka, Nanoscale, 2010, 2, 1657–1660.
- 17 J. Bursik, M. Soroka, R. Uhrecky, R. Kuzel, F. Mika and S. Huber, Appl. Surf. Sci., 2016, 376, 209–218.
- 18 J. Lu, S. Cheng, M. Liu and C. Jia, CrystEngComm, 2016, 18, 8038–8043.
- 19 A. Matsuda, R. Yamauchi, D. Shiojiri, G. Tan, S. Kaneko and M. Yoshimoto, Appl. Surf. Sci., 2015, 349, 78–82.
- 20 J.-M. Li, CrystEngComm, 2017, 19, 3392–3397.
- 21 J.-M. Li, Cryst. Growth Des., 2009, 9, 4171–4175.
- 22 M. N. R. Ashfold, F. Claeyssens, G. M. Fuge and S. J. Henley, Chem. Soc. Rev., 2004, 33, 23–31.
- 23 H. M. Christen and G. Eres, J. Phys.: Condens. Matter, 2008, 20, 264005.
- 24 A. A. Belik, S. Iikubo, K. Kodama, N. Igawa, S. Shamoto, S. Niitaka, M. Azuma, Y. Shimakawa, M. Takano, F. Izumi and E. Takayama-Muromachi, Chem. Mater., 2006, 18, 798– 803.
- 25 L. Qiao, H. Y. Xiao, H. M. Meyer, J. N. Sun, C. M. Rouleau, A. A. Puretzky, D. B. Geohegan, I. N. Ivanov, M. Yoon, W. J. Weber and M. D. Biegalski, J. Mater. Chem. C, 2013, 1, 4628–4633.
- 26 H. Bea, M. Bibes, A. Barthelemy, K. Bouzehouane, E. Jacquet, A. Khodan, J. P. Contour, S. Fusil, F. Wyczisk, A. Forget, D. Lebeugle, D. Colson and M. Viret, Appl. Phys. Lett., 2005, 87, 072508.
- 27 S. Fujino, M. Murakami, S. H. Lim, M. Wuttig, L. G. Salamanca-Riba and I. Takeuchi, Solid State Ionics, 2007, 178, 1257–1261.
- 28 L. W. Martin, Y. H. Chu and R. Ramesh, Mater. Sci. Eng., R, 2010, 68, 89–133.
- 29 Y. Uratani, T. Shishidou, F. Ishii and T. Oguchi, Jpn. J. Appl. Phys., 2005, 44(Part 1), 7130–7133.