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Mesoporous TiO<sub>2</sub> hierarchical structures: preparation and efficacy in solar cells†

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We investigated an electrospray-based method to manufacture photoelectrodes for dye-sensitized solar cells (DSSCs).  $TiO_2$  doughnut-, spherical-, and disk-shaped particles with a large surface area, high crystallinity, uniform nanostructure, and good light scattering properties were fabricated via a simple electrospray method. The control of the morphology of the nanostructured particles prepared by electrospraying a dispersion of nanoparticles was investigated experimentally; the results are qualitatively explained on the basis of the available theory. The solvent in the droplet, droplet size, surface tension, process temperature, and process humidity are crucial to the morphology of the resulting particles. The effect of the particle morphology on the performance of DSSCs is demonstrated. Compared to the DSSCs with conventional photoelectrodes (with a power conversion efficiency of 8.4%), the DSSCs based on doughnut-, spherical-, and disk-shaped particle photoelectrodes yielded higher power conversion efficiencies of 8.8%, 9.3%, and 10.4%, respectively. The DSSC utilizing the disk-type photoelectrode showed the best performance under AM1.5 global illumination through a photo-mask at an illumination power of 100 mW cm $^{-2}$ . This is because the generated  $TiO_2$  disks provide a large surface area and exhibit excellent light scattering capabilities, thus resulting in a low total internal resistance and long electron lifetime.

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### 1. Introduction

The advantages of simple fabrication using low cost materials and high energy conversion efficiency make dye-sensitized solar cells (DSSCs) very attractive candidates for next generation solar energy conversion technologies.<sup>1-3</sup> In order to improve the efficiency of DSSCs, various techniques such as deposition of a thin tunneling barrier layer on the substrate<sup>4,5</sup> or the oxide surface,<sup>6</sup> co-sensitization using different dyes,7 post-treatment with a TiCl4 precursor, 8,9 and use of reflection layers 10 have been proposed. Light-scattering particulate films with a bilayer structure have also been adopted widely because they confine the incident light within an electrode and thereby enhance the photocurrent density.11-15 The size of the scattering particle should be large enough to scatter the incident light efficiently since small-sized (10-30 nm) nanocrystalline particles cause Rayleigh scattering that is too weak to scatter the light backward.16 Large TiO2 particles with spherical shapes or flat surfaces are normally used as scattering particles. However, except for their role in the actual light scattering, these scattering particles hardly offer any additional effect, for instance electron generation, because dye adsorption is expected to be much less for these particles than

submicrometer-sized aggregates assembled by nanocrystallites appear to be a promising approach to meet such conflicting requirements. Hierarchically-structured TiO2 (HS-TiO2) has been suggested as the most promising photoelectrode for DSSCs because of the advantages of (1) large surface area and dye adsorption; (2) light scattering effects of the submicrometer-sized structures; and (3) better penetration of electrolytes through the relatively large pores between the HS-TiO2 particles. 19-21 Recently, several groups have reported that the morphology of hierarchical structured TiO2 spheres is optimal for DSSCs. 19,20,22-26 However, it is not simple to obtain such hierarchical structures using multistep processes such as the sol-gel method,22 solvothermal reactions,27 or hydrolysis.19 These processes are neither practical nor cost-effective for the large-scale production of DSSC modules. The electrospray technique has recently evolved as a cheap and simple process to directly deposit thin films from their colloidal solutions. The film size and uniformity of TiO2 photoelectrodes could be easily controlled by varying the number of nozzles, volume of the syringe pump, and the motion-controlled system with a micro-processor. Photographs of the 10  $\times$  10 cm and 30  $\times$ 30 cm films are shown in the ESI (Fig. S1†). This technique can be applied widely in modern material technologies such as microelectronics and nanotechnology28 and in industries for the deposition of various ceramic powders, polymer powders,29 and TiO<sub>2</sub> electrodes for DSSCs.30,31

that for nanocrystalline TiO2 owing to their much smaller surface

area.17,18 The use of hierarchically structured films having

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In this paper, we report an innovative particle processing technique to produce electrosprayed nanocrystalline-TiO2 (nc-TiO<sub>2</sub>) with several morphologies (spherical, biconcave, doughnut, and disk) for DSSC photoelectrode. A comparison between a theoretical mechanism and the current experimental results is shown to clarify how the particles of various sizes and morphologies are formed. When compared to the traditional photoelectrodes, the present electrosprayed doughnut, spherical-, and disk-shaped particle photoelectrodes showed several advantages. Firstly, the electrosprayed TiO2 photoelectrode particles were composed of nanocrystallites with diameters of approximately ~15-20 nm, and thus, they did affect the dyeloading capacity of the photoelectrodes adversely. Secondly, the diameter of the electrosprayed TiO2 photoelectrodes particles was submicron-sized, resulting in much higher diffuse reflection properties, thereby effectively improving the light harvesting characteristics of the device. Thirdly, morphology of the photoelectrodes could be easily engineered and fine-tuned by changing the operating parameters of the electrospraying process.

## 2. Experimental section

#### Preparation of various TiO2 particles

First, 10 wt% of nc-TiO<sub>2</sub> (Degussa P-25, particle size  $\sim$ 15–20 nm) was dispersed in ethanol using a Kotobuki UAM-015 Ultra Apex mill. The dispersion of 10 wt% nc-TiO2 in ethanol was then dispersed in an ethanol/butanol (9/1 v/v) solution. The diluted 5 wt% nc-TiO2 dispersion solution was loaded into a plastic syringe, which was connected to a Bertan 205B high-voltage power supply. The dispersed nc-TiO2 solution was electrosprayed directly onto the conducting fluoridated tin oxide (FTO) substrates having the dimensions of 10 cm imes 10 cm. The process temperature and humidity were controlled by a heater and humidifier, respectively. Table 1 lists the temperature and humidity values used to produce different particle morphologies in this experiment. An electric field of 15 kV was applied between the metal orifice and the conducting substrates, which were separated by a distance of 10 cm. The feed rate was controlled by a syringe pump at 35-30 µL min<sup>-1</sup>. In order to obtain a uniform thickness over a large area, the nozzle and the

Table 1 nc-TiO $_2$  dispersion solvent and processing conditions. Temperature and humidity were controlled by a heater and humidifier, respectively. The boiling points of ethanol and butanol are 78.4 and 117.9 °C, respectively

Morphology	Solution	Temperature (°C)	Humidity (%)
Sphere	5 wt% nc-TiO <sub>2</sub> (P-25)	25	20
Biconcave	in ethanol 5 wt% nc-TiO <sub>2</sub> (P-25)	25	20
Doughnut	in ethanol/butanol (9/1 v/v) 5 wt% nc-TiO <sub>2</sub> (P-25)	35	20
· ·	in ethanol	33	20
Disk	5 wt% nc-TiO $_2$ (P-25) in ethanol	25	40

substrate were placed on a motion-control system with a microprocessor. The surface areas of the resulting nanoparticles were determined by the Brunauer-Emmett-Teller (BET) method; the distributions of the pore volume and particle size were determined by the Barrett-Joyner-Halenda (BJH) method using the adsorption branches of the N<sub>2</sub> isotherms. Dye desorption experiments were carried out using the nanoparticle photoelectrode films coated with the N719 dye prepared from a 0.3 mM ethanolic solution. For each sample, the dye was desorbed from the nanoparticle photoelectrode by treatment with a certain volume of 0.1 M NaOH (aq). The absorbance of the resulting solution was then quantified with a UV-vis spectrophotometer by calculating the number of molecules from the obtained UV-vis absorption spectra of the desorbed sensitizers using the extinction coefficient of the N719 sensitizer. The optical properties were measured by a UV/VIS/NIR spectrometer (Perkin Elmer LAMBDA 1050) within the wavelength range of 185–3300 nm. The surface and cross-sectional morphologies of the TiO<sub>2</sub> nanoparticles were analyzed by field-emission transmission electron microscopy (FE-TEM) (Hitachi, HF-3300) and field-emission scanning electron microscopy (FE-SEM; Hitachi, S-4800). The structural properties of the TiO<sub>2</sub> nanoparticles were analyzed by X-ray diffraction (XRD; PANalytical, X'pert PRO-MPD with Cu K $\alpha$  irradiation) in the  $\theta$ -2 $\theta$  mode.

#### **DSSC** fabrication

To fabricate a conventional DSSC, an nc-TiO<sub>2</sub> paste (Solaronixk T/SP, R/SP) was applied to an FTO-glass slide using a doctor blade and subsequently heat treated. Cellophane tape was used to control the film thickness. To produce the final structure, a conventional scatter layer was pasted onto the nc-TiO2 layer by a doctor blade. After the electrospray process, each as-prepared TiO2-coated FTO photoelectrode was placed between two planar steel plates for mechanical pressing with a Teflon film to prevent the delamination of the TiO2 photoelectrodes. Mechanical hot-pressing process can indeed improve the adhesion between the TiO2 photoelectrode and the substrate, thus improving the interfacial performance of TiO<sub>2</sub> photoelectrodes in DSSCs. The typical pressure applied to fabricate the solar cells was 12 MPa at 120 °C for 10 min. The photoelectrodes coated with the TiO<sub>2</sub> spherical-, doughnut-, and disk-shaped particles were sintered under air at 500 °C for 30 min. The TiO<sub>2</sub> biconcave particles separated from the substrate by peeling after the mechanical pressing process. This might have resulted because of the hollow particle structures, as shown in Fig. 1. Therefore, the device performance could not be tested for the biconcave particle morphology. For post-treatment, the TiO<sub>2</sub> films were soaked in a 0.05 M TiCl<sub>4</sub> aqueous solution for 20 min at 80 °C, rinsed with water, and then sintered again at 450 °C for 30 min. After cooling to 80 °C, the TiO<sub>2</sub> photoelectrodes were immersed in a purified  $3 \times 10^{-4}$  M *cis*-diisothiocyanato-bis(2,2'-bipyridyl-4,4'-dicarboxylato)ruthenium(II)bis(tetrabutylammonium) solution (N719, Solaronix) for 15 h at room temperature (25 °C). For the counter electrode, the FTO plates were drilled by a microdrill, washed with a 0.1 M HCl solution in ethanol, and subsequently cleaned in an ultrasonic bath with water and ethanol for

Sphere particles Solvent evaporation Surface Shell/Skin Changed solven Biconcave particles Increased Doughnut particles temperature Mushroom Inorganic material/solution Increased humidity Disk particles Convey Disk

Fig. 1 Formation mechanisms for various particle morphologies. R1: sphere, R2: biconcave, R3: doughnut, and R4: disk.

15 min. A solution of 5 mM  $H_2PtCl_6$  in isopropyl alcohol was drop-cast onto the washed FTO plates and then sintered at 400 °C for 20 min under air. The dye-adsorbed  $TiO_2$  photoelectrodes were rinsed with ethanol and dried under nitrogen flow. The dye-adsorbed  $TiO_2$  photoelectrodes were then assembled and sealed with the Pt counter electrodes using thermal adhesive films (Surlyn, Dupont 1702, 25  $\mu$ m-thick) as spacers to produce sandwich-type cells. The liquid electrolyte consisted of 0.7 M 1-propyl-3-methylimidazolium iodide (PMII), 0.03 M iodine ( $I_2$ ), 0.1 M guanidinium thiocyanate (GSCN), and 0.5 M 4-tert-butyl-pyridine (TBP) in a mixture of acetonitrile (ACN) and valeronitrile (VN) (85/15 v/v). The electrolyte solution was introduced through the holes drilled in the counter electrodes. Finally, the holes were sealed with a hot-melt film and a cover glass.

#### Photovoltaic and photoelectrical measurements

The photovoltaic measurement of the DSSCs employed an AM1.5G solar simulator between the sample and a 450 W xenon lamp. The intensity of the simulated light was calibrated by a silicon reference solar cell equipped with a KG5 filter for approximating AM1.5 global solar radiation. The photovoltaic characteristics of the DSSCs were obtained by applying an external potential bias to the cells and measuring the generated photocurrent with a Keithley Model 2400 source meter. The incident photon-to-electron conversion efficiency (IPCE) was measured as a function of the wavelength from 350 to 800 nm using a specially designed IPCE system for DSSCs (PV measurement, Inc.). A 75 W xenon lamp was used as the light source to generate a monochromatic beam. The calibration was performed using a NIST calibrated silicon photodiode as the standard. The IPCE values were collected at a low chopping speed of 5 Hz. The electrical impedance spectra (EIS) were measured using an impedance analyzer (Solatron 1260) at an open-circuit potential under an AM1.5G full-sun illumination (100 mW cm<sup>-2</sup>) at frequencies in the range of  $\sim$ 0.1–10<sup>5</sup> Hz. The

obtained spectra were fitted using ZView software (v3.2c, Scribner Associates, Inc.).

Side view

#### Results and discussion

Fig. 1 shows the formation mechanisms for different particle morphologies using the electrospray method. Various particle morphologies could be produced by changing the processing conditions and the initial droplet shapes. A modified Peclet number (Pe) can be used to characterize the morphology of nanoparticles fabricated using the electrospray process.<sup>32</sup>

Based on this idea, Pe value can be used to determine the solvent evaporation rate, the particle diffusion coefficient within the droplet, and the particle diffusion rate. This relationship is similar to that suggested by Okuyama and Wuled Lenggoro<sup>33</sup> and can be written as

$$Pe = \frac{\frac{\partial r_{\rm d}}{\partial t} r_{\rm d}}{D_{\rm AR}} \tag{1}$$

where  $\partial r_{\rm d}/\partial t$  is the rate of change of the droplet radius because of the solvent evaporation,  $r_d$  is the droplet radius,  $D_{AB}$  is the diffusion coefficient of the nanoparticle solute inside the droplet, and Pe is the ratio of the solvent evaporation rate from the droplet surface and the diffusion rate of the nanoparticles inside the droplet. The electrospray method typically yields spherical particles. We could easily fabricate spherical particles using the nc-TiO<sub>2</sub> dispersion solution (Fig. 1, R1). The solution exhibits the maximum structural stability when the particles are in spherical form. This is the fundamental reason for the production of this shape.34 When the humidity is low, the solvent evaporation rate is high, and the nanoparticles can rehomogenize as droplet shrinks, promoting the formation of less porous particles. The low porosity corresponds to a high mechanical strength inside the particles, so the particles do not experience shrinkage or collapse, and a spherical morphology is **RSC Advances** Paper

formed. For the same droplet shrinkage rate, the Pe value is the lowest in this case because of the small droplet radius.<sup>32</sup>

However, changing this parameter, which is related to the solvent evaporation rate, can result in the formation of different particle morphologies.34 When the solvent evaporation rate was slow because of the use of the ethanol/butanol (9/1 v/v) solvent mixture (boiling points of 78.4 and 117.7 °C, respectively), as shown in Table 1, the removal of the solvent from the droplet surface caused an increase in the concentration of nanoparticles near the surface.

As more of the solvent evaporated, the concentration increased until a thin crust of solid nanoparticles was formed at the droplet surface. As more solvent was removed, the thickness of the crust increased until finally a large hollow particle was formed; this hollow shell collapsed to form the observed biconcave particle (Fig. 1, R2). At this droplet shrinkage rate, the Pe value was high in this case. The particle was spherical at low values of Pe and became biconcave when the value of Pe increased.32

A toroid morphology of the produced particles can be considered to arise from an initial deformation of the droplet, which changed the initial spherical shape to either a doughnutlike shape or a disk form. This is possible by the loss of structural stability of the droplet in a two-phase flow field because of the additional macro- and microhydrodynamic effects during the drying process.35

The structural stability of droplets can be explained by the bond's number  $\beta$ , the ratio of the inertial force to the surface tension effects. The bond number is given by

$$\beta = \frac{\Delta \rho a d_{\rm d}^2}{\sigma} \tag{2}$$

where  $\Delta \rho$  is the difference in the densities of the droplet and the surrounding fluid, a is the acceleration (change in gas velocity with time),  $d_{\rm d}$  is the droplet size, and  $\sigma$  is the surface tension. <sup>36,37</sup> The droplet is nearly spherical when  $\beta \rightarrow 0$  and becomes flat when the value of  $\beta$  increases. The value of  $\beta$  can be set experimentally by varying the process temperature or humidity. It follows that the destabilization of the initial shape of a droplet could be achieved by (i) increasing the droplet size and (ii) decreasing the surface tension ( $\sigma$ ) of the droplet. A decrease in  $\sigma$ by increasing the process temperature to  $\sim$ 35 °C using a heater, as shown in Table 1, destabilizes the droplet. For droplets at high temperatures, the values of  $\beta > 1$  are attained in eqn (2), and a doughnut-like form is obtained (Fig. 1, R3). At high values of humidity ( $\sim$ 40%) obtained by using a humidifier, the droplet radius increases with the decreasing solvent evaporation rate, and the destabilization of the initial droplet shape occurs. The values of  $\beta > 1$  are also attained from eqn (2), and the disk form is obtained (Fig. 1, R4). Detailed formulae for the determination of shape in all the cases are available in previous reports. 37-40 The selection and adjustment of the initial physicochemical properties of the droplet is known as the bond-number correlation. The changes in the hydrodynamics, structural stability, and droplet behavior cause the transformation of the shape of a droplet from spherical to doughnut or disk-like (Fig. 1, R3 and R4).35

The effectiveness of the above mentioned mechanism for fabricating particles with various morphologies is discussed using the SEM images shown in Fig. 2(a-d).

Fig. 2a shows the spherical TiO<sub>2</sub> particles produced using the electrospray method. The SEM images confirm that under typical conditions, the electrospray method generates spherical particles.41 The adjustment of some experimental parameters produces particles with different morphologies. The change in the solvent evaporation rate causes the production of biconcave particles (Fig. 2b), which is attributed to the component movements within the droplet42 and to the decreased

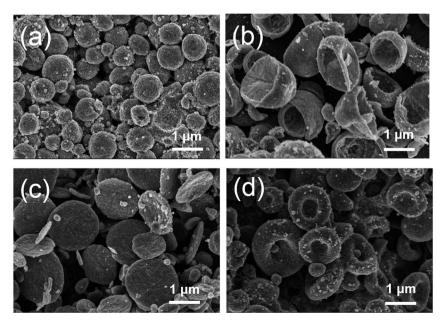


Fig. 2 SEM images of particles of (a) spherical, (b) biconcave, (c) disk -, and (d) doughnut -like morphologies.

(a) Disk (b) Doughnut (C) Sphere

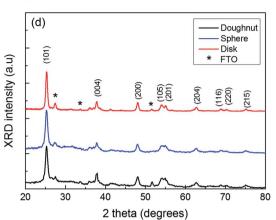


Fig. 3 TEM images of (a) disk-, (b) doughnut-, and (c) sphere-shaped  ${\rm TiO_2}$  particles; (d) XRD diffraction patterns of the  ${\rm TiO_2}$  doughnut-, sphere-, and disk-shaped particles prepared by electrospraying on FTO substrate.

evaporation rate of the mixed solvent. In the other cases, the changes in the processing conditions related to the hydrodynamics, structural stability, and droplet behavior cause the formation of doughnut- and disk-shaped particles (Fig. 2c and d, respectively).<sup>35</sup> While the morphology of the electrosprayed particles prepared using a TiO<sub>2</sub> dispersion solution is generally spherical, the existence of biconcave morphology has also been reported in previous publications; however, the existence of doughnut and disk morphologies has not been reported. Here, we report the formation and observation of these shapes for the first time.

Fig. 3(a)–(c) show the TEM images of the electrosprayed  $TiO_2$  particles. The electrosprayed disk-, doughnut-, and spherical-shaped  $TiO_2$  particles are composed of  $\sim$ 15–20 nm P-25 (Degussa) nanoparticles. The crystallite sizes of disk-, spherical-, and doughnut-shaped  $TiO_2$  particles, as determined from the anatase (101) peak using the Scherrer equation, were 15.3, 16.8, and 17.2 nm, respectively, all of which are comparable to the particle sizes obtained from the TEM images. Furthermore, Fig. 3(d) shows the XRD patterns for the  $TiO_2$  sphere, doughnut, and disk particles, which indicate that all the particles possessed highly crystalline anatase structures without any impurity phases (ICSD 156838). Other peaks observed at 26, 33, and 51° are attributed to the FTO substrate (ICSD 039176).

The physical properties of the films bearing the disk-, spherical-, doughnut-shaped, and conventional particles are given in Table 2.

The surface area and pore volume of the disk-, spherical-, doughnut-shaped, and conventional particles were found to be 99.5 m $^2$  g $^{-1}$  and 0.402 cm $^3$  g $^{-1}$ , 90.9 m $^2$  g $^{-1}$  and 0.394 cm $^3$  g $^{-1}$ , 89.7 m $^2$  g $^{-1}$  and 0.374 cm $^3$  g $^{-1}$ , and 82.3 m $^2$  g $^{-1}$  and 0.353 cm $^3$  g $^{-1}$ , respectively. The mesoporous TiO<sub>2</sub> hierarchical structures

**Table 2** The physical properties of the films bearing the disk-, spherical-, doughnut-shaped, and conventional particles

Morphology	Surface area (m² g <sup>-1</sup> )	Pore volume (cm <sup>3</sup> g <sup>-1</sup> )	Absorbed dye $(10^{-7} \text{ mol cm}^{-2})$
Disk	99.5	0.402	4.53
Sphere	90.9	0.394	4.35
Doughnut	89.7	0.374	4.12
Conventional	82.3	0.353	3.98

were annealed under high temperature at 500 °C for 30 min. After annealing, these structures were soaked in a TiCl<sub>4</sub> aqueous solution for 20 min at 80 °C for post-treatment and then sintered again at 450 °C for 30 min. It is well known that post-treatment by a TiCl<sub>4</sub> aqueous solution and the hightemperature annealing process both improve connections between the nanoparticles by changing the ratio of surface to interface. Post-treatment with TiCl4 aqueous solution also changes the surface area by modifying the roughness factor of the nanoparticles. Consequently, these post-treatments can change the surface area and interface; these changes can strongly affect the disk-, spherical-, and doughnut-shaped TiO<sub>2</sub> hierarchical structures because the different assembly patterns produce varied pore sizes and interfaces. Therefore, we can assume that the surface area is different for each type of mesoporous TiO<sub>2</sub> hierarchical structure, despite the constant primary unit of the samples of P-25. The amount of the dye adsorbed on the disk-, spherical-, doughnut-shaped, and conventional particles was  $4.53 \times 10^{-7}$ ,  $4.35 \times 10^{-7}$ ,  $4.12 \times 10^{-7}$  $10^{-7}$ , and  $3.98 \times 10^{-7}$  mol cm<sup>-2</sup>, respectively. The pore-size distributions for these four films were analyzed by N2 adsorption-desorption isotherms. As shown in Fig. S2,† the average pore size of the film from conventional TiO2 NPs was 31 nm, a reasonable result considering the constituent particle size. Conspicuously, two kinds of pores were observed in the films fabricated from the disk-, spherical-, and doughnutshaped  $TiO_2$  structures. Pores of  $\sim$ 26-30 nm and  $\sim$ 120-160 nm were found in the films derived from TiO2 hierarchically structured particles. The  $\sim$ 26-30 nm-sized pores were the internal pores of the nanoporous disk-, spherical-, and doughnut-shaped TiO<sub>2</sub> structures, whereas the  $\sim$ 120–160 nmsized pores originated from the interstitial voids formed by the closed-packed sub-micrometer TiO2 hierarchical structures. These pores allowed the electrolyte to penetrate through the TiO<sub>2</sub> film. The HS-TiO<sub>2</sub> particles exhibited a high surface area, high pore volume per gram, and high dye-loading as compared to those exhibited by the conventional TiO<sub>2</sub> particles.

Fig. 4 shows the cross-sectional images of the electrosprayed electrodes ( $\sim$ 15  $\mu$ m thick) with uniform spherical, doughnut, and disk-like shapes from the bottom to the top of the deposited layers. The photovoltaic properties of the DSSCs formed using each of the TiO<sub>2</sub> photoelectrodes are shown in Fig. 5(a) and Table 3. The DSSC with the conventional photoelectrode showed a short circuit current ( $J_{sc}$ ) of 14.7 mA cm<sup>-2</sup>, an open circuit voltage ( $V_{oc}$ ) of 0.805 mV, and a fill factor of 71%, resulting in a power conversion efficiency (PCE) of 8.4%. The

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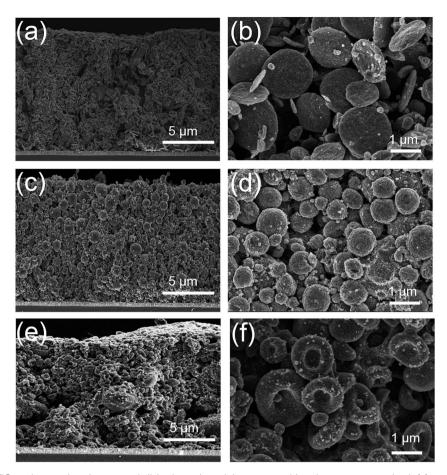


Fig. 4 SEM images of TiO<sub>2</sub> sphere-, doughnut-, and disk-shaped particles prepared by electrospray method. (a) and (b) Cross-sectional and surface images, respectively, of the disk-shaped particles deposited on FTO substrate; (c) and (d) show similar cross-section and surface images for spherical particles; (e) and (f) show cross-sectional and surface images for doughnut-shaped particles.

DSSC with the photoelectrode consisting of doughnut-shaped particles showed a  $J_{sc}$  of 15.0 mA cm<sup>-2</sup>,  $V_{oc}$  of 0.817 mV, and fill factor of 71%, resulting in a PCE of 8.8%. The DSSC with the photoelectrode consisting of spherical particles showed an enhanced  $J_{\rm sc}$  of 15.7 mA cm<sup>-2</sup>,  $V_{\rm oc}$  of 0.820 mV, and fill factor of 72%, resulting in an increased PCE of 9.3%. The DSSC with the photoelectrode consisting of disk-shaped particles showed a significantly enhanced  $J_{sc}$  of 17.5 mA cm<sup>-2</sup>,  $V_{oc}$  of 0.825 mV, and fill factor of 72%, resulting in an increased PCE of 10.4%. The DSSC utilizing the disk-type photoelectrode showed the best performance with an active area of 0.25 cm<sup>2</sup>, as shown in Table 3. In particular, the  $J_{\rm sc}$  was 11%, 16%, and 19% greater than that of the DSSCs using the spherical-, doughnut-shaped, and conventional photoelectrodes, respectively. Statistical analyses of the PCE values of the devices based on TiO<sub>2</sub> photoelectrodes from disk-, spherical-, doughnut-shaped, and conventional particles are shown in the ESI (Fig. S3†). Data was collected from more than 20 devices fabricated using each type of shaped particle. This is in good agreement with the dyeloading results discussed above. The fill factor was almost identical among the three photoelectrode types.

The nanoparticle-coated photoelectrodes for the DSSCs appeared semi-transparent, suggesting the transmittance of a significant portion of the visible light. It is well-known that an

increase in the light-scattering ability of TiO2 photoelectrodes leads to an increase in the light-harvesting efficiency of the dyed film, and thereby to an increase in the incident photon-tocurrent conversion efficiency (IPCE) of the pertinent DSSC. To investigate the scattering behavior of the TiO<sub>2</sub> photoelectrodes, we measured the diffuse reflectance values of the TiO2 photoelectrodes with doughnut-, spherical-, disk-shaped, and conventional nanoparticles by using the integrating spheremode of a UV-spectrophotometer, as shown in Fig. 5(b). Throughout the wavelength region ( $\lambda = 400-800$  nm), the disktype photoelectrode maintains the highest reflectance (R) compared to the spherical-, doughnut-type, and conventional electrodes, indicating the superior light-scattering properties of the disk-type film. The enhanced light reflectance can be attributed to the unique planar structure of the disk-type film. Diffusive particles randomly scatter in all directions. Thus still much of the incident light is lost after the scattering layer. However, disk-type film with the planar structure can directly reflect a large fraction of light back into the devices. It is expected such a planar structure can significantly enhance light harvesting and improve the overall conversion efficiency. 43,44

In order to compare the photovoltaic performance of the disk-type photoelectrode to that of the electrode using spherical-, doughnut-shaped, and conventional particles, the IPCE

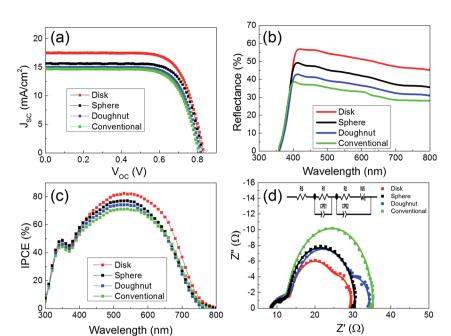


Fig. 5 (a) Photovoltaic parameters of the DSSCs using  $TiO_2$  photoelectrodes containing disk-, spherical-, doughnut-shaped, and conventional particles, (b) reflectance spectra of the DSSCs with  $TiO_2$  photoelectrodes of disk-, spherical-, doughnut-shaped, and conventional particles, (c) IPCE curves of the DSSCs with  $TiO_2$  photoelectrodes of disk-, spherical-, doughnut-shaped, and conventional particles, and (d) EIS of the DSSCs with  $TiO_2$  photoelectrodes of disk-, spherical-, doughnut-shaped, and conventional particles.

Table 3 Photovoltaic properties of the DSSCs assembled using photoelectrodes comprising thin films of disk-, spherical-, doughnut-shaped, and conventional  $nc-TiO_2$  particles

Morphology	$V_{oc}(V)$	$J_{sc} \left( \text{mA cm}^{-2} \right)$	FF (%)	PCE (%)
Disk	0.825	17.5	72	10.4
Sphere	0.820	15.7	72	9.3
Doughnut	0.817	15.0	71	8.8
Conventional	0.805	14.7	71	8.4

spectra of the  ${\rm TiO_2}$  photoelectrodes using these materials were collected. As shown in Fig. 5(c), the photoelectrode prepared using the disk-shaped particles possessed higher IPCE values over a wide spectral range (from 400 to 750 nm) than a photoelectrode of similar thickness using the spherical, doughnut-shaped, or conventional particles. The higher IPCE values of the disk photoelectrode are ascribed to the enhanced light-scattering capacity of the photoelectrode, which promotes the light harvesting of the sensitizing dye in this region. This is in good agreement with the  $J_{\rm sc}$  and reflectance results discussed above.

In order to investigate the electron transport properties in the  $\operatorname{nc-TiO}_2$  doughnut-, spherical-, disk-shaped, and conventional photoelectrodes, the EIS measurements were employed. Fig. 5(d) shows the Nyquist plots of the EIS of the disk-, doughnut-, and spherical-type devices, measured under 1.5 AM simulated solar conditions. Generally, the conventional three semicircles observed in the Nyquist plot indicate the internal resistances of the devices. The series resistance ( $R_0$ ) of the cells is estimated from the intercept on the real axis with the highest

frequency; the first arc in the high-frequency range indicates the electrochemical reaction resistance or redox couple at the Pt counter electrode  $(R_1)$ ; the second arc in the intermediate-frequency range demonstrates the charge recombination resistance at the interface between the  $\text{TiO}_2$  photoelectrode and the electrolyte  $(R_2)$ ; and the third arc is caused by the Warburg diffusion of the redox species  $(W_{\rm S})$ . <sup>46</sup> CPE<sub>1</sub> and CPE<sub>2</sub> are the constant-phase elements of the capacitance, corresponding to  $R_1$  and  $R_2$ , respectively. <sup>47</sup>

Fig. 5(d) illustrates the origin of the differences in the electric properties between the devices using doughnut-, spherical-, disk-shaped, and conventional  $TiO_2$  photoelectrodes. As shown in Table 4, the disk photoelectrode had a smaller  $R_2$  resistance and larger  $CPE_2$  capacitance than the spherical-, doughnut-shaped, and conventional photoelectrodes.

This result implies that the DSSC with the disk photoelectrode had more efficient charge-transfer at the TiO<sub>2</sub>/electrolyte interface. The packing density is rather low due to the point attachment of spherical scattering particles, which is adverse to light harvesting and electron transfer. Therefore, a special scattering layer constructed from planar mesostructured particles with high surface area and sufficient attachment to each other is expected to be one of the good choices to improve electron transfer and light harvesting for benefiting cell performance. Moreover, it has an improved electron lifetime, as determined by the following formula:<sup>30</sup>

$$\tau_n = R_2 \times \text{CPE}_2 \tag{3}$$

The calculated lifetimes  $(\tau_n)$  of the disk-, spherical-, doughnut-shaped, and conventional photoelectrodes are 15.2,

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Table 4 Parameters of the DSSCs obtained from the fitted EIS curves using the equivalent circuit shown in Fig. 5(d)

Morphology	$R_{0}\left( \Omega  ight)$	$R_1\left(\Omega\right)$	$CPE_1$ (mF)	$R_2\left(\Omega\right)$	$CPE_2$ (mF)	$W_{ m S}\left(\Omega ight)$	τ (ms)
Disk	8.26	5.69	1.03	11.72	1.3	3.91	15.2
Sphere	8.59	5.69	0.92	14.33	0.89	3.14	12.7
Doughnut	8.41	5.93	0.7	14.93	0.8	4.45	11.9
Conventional	8.58	5.08	0.93	15.45	0.28	6.72	4.32

12.7, 11.9, and 4.3 ms, respectively. Under illumination, the relation between  $V_{oc}$  and recombination lifetime can be described as:

$$V_{\text{oc}} = \frac{RT}{\beta F} \ln \left( \frac{AI}{n_0 k_1 \left( \mathbf{I}_3^- \right) + n_0 k_2 \left( \mathbf{D}^+ \right)} \right) \tag{4}$$

where R is the molar gas constant, T is the temperature, F is the Faraday constant,  $\beta$  is the reaction order for  $I_3^-$  and electrons, A is the electrode area, I is the incident photon flux,  $n_0$  is the concentration of electronic states in the conduction band,  $k_1$  is the reaction constant for the back-reaction of injected electrons with triiodide, and  $k_2$  is the reaction constant for the recombination of electrons with oxidized dye (D+).48 If we assume that recombination with the oxidized dye molecules is negligible, then  $V_{oc}$  becomes logarithmically dependent on  $k_1$  under the conditions used in this study because the concentration of triiodide in the electrolyte and the incident photon flux were both constant. Furthermore, if  $V_{\rm oc}$  is proportional to  $\ln(1/\tau_n)$ , then this suggests that the DSSCs with longer lifetimes have higher  $V_{\rm oc}$  values. In particular, the long lifetime of the diskshaped photoelectrode should increase the  $V_{\rm oc}$  to 0.825 V, compared to the  $V_{oc}$  values of 0.820, 0.817, and 0.805 V for the doughnut-shaped, and conventional electrodes, respectively. The increased lifetime for the disk photoelectrode was directly related to the reduced possibility of multiple trapping, as well as the inhibition of the recombination reactions with the electrolyte.<sup>49</sup> Finally, the  $W_S$  value of the disk photoelectrode (3.9  $\Omega$ ) showed a smaller diffusion resistance than that of the doughnut  $(4.5 \Omega)$  and conventional  $(6.7 \Omega)$ photoelectrodes. This suggests that the disk photoelectrode had a good pore structure and allowed an efficient diffusion of the I/ I<sub>3</sub> redox couple in the electrolyte as shown in Table 2.46 A threedimensional network the interconnected pores in the disk photoelectrode promoted the electrolyte diffusion and reduced the diffusion resistance.

#### 4. Conclusion

In summary, hierarchically structured doughnut-, spherical-, and disk-shaped particles were successfully formed using the electrospray process in this study. Comparisons between a theoretical mechanism and experimental results were shown to illustrate how the particles of various morphologies were formed. The adjustment of some experimental parameters produced particles with different morphologies. The change in the solvent evaporation rate caused the production of biconcave particles, which is attributed to the component movements

within the droplet and to the decreased evaporation rate of the mixed solvent. In the other cases, the changes in the processing conditions related to the hydrodynamics, structural stability, and droplet behavior caused the formation of doughnut and disk particles. Electrosprayed TiO<sub>2</sub> doughnut-, spherical-, and disk-shaped particles can be used as effective photoelectrodes in DSSCs to improve the performance of the device. Compared the conventional photoelectrode, the electrosprayed doughnut-, spherical-, and disk-shaped photoelectrodes consisted of numerous nanocrystallites ( $\sim$ 15-20 nm size) packed on the particles. This induces the light scatter effect with an added advantage of high dye-loading capacity. The disk-shaped photoelectrode had a higher diffuse reflectance capacity than those of the doughnut, spherical-shaped, and conventional photoelectrodes in the wavelength range between 400 and 800 nm, indicating superior light-scattering properties of the disk-shaped photoelectrodes. Thus, our unique configuration with doughnut-, spherical-, and disk-shaped photoelectrodes can improve the light harvesting efficiency of DSSCs without compromising the dye loading abilities. High efficiencies of 10.4%, 9.3%, and 8.8% could be achieved with the disk-, spherical-, and doughnut-shaped photoelectrodes, respectively, which are 24%, 11%, and 5% greater than that of a device with the conventional nc-TiO<sub>2</sub> particles photoelectrodes having only 8.4% efficiency.

#### Conflicts of interest

There are no conflicts to declare.

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