



Cite this: *Chem. Soc. Rev.*, 2018, 47, 4279

Excited-state structural relaxation and exciton delocalization dynamics in linear and cyclic π -conjugated oligothiophenes

Kyu Hyung Park,^a Woojae Kim,^a Jaesung Yang^b and Dongho Kim *^a

π -Conjugated oligothiophene is considered a chain segment of its polymeric counterpart, whose size and shape can be precisely controlled. Because of its simplified structure, it is possible to understand complex excited-state dynamics of the π -conjugated polymers by employing a bottom-up approach. We review theoretical and experimental aspects of π -conjugated oligothiophenes by summarizing recent works employing time-resolved spectroscopy. The extent of exciton delocalization, which is a prerequisite to efficient charge generation at organic heterojunctions, is described sequentially in model linear and cyclic oligothiophenes, and their analogues. The heterogeneous nature of these systems is highlighted by illustrating the results at both ensemble and single-molecule levels. Exciton dynamics that arise in the polymers are also covered and the significance of exciton and charge delocalization in photovoltaic materials is highlighted.

Received 16th August 2017

DOI: 10.1039/c7cs00605e

rsc.li/chem-soc-rev

Key learning points

- (1) Electronic structures of linear and cyclic π -conjugated oligomers.
- (2) Relationship between absorption/emission characteristics and the extent of exciton delocalization.
- (3) Excited-state dynamics in π -conjugated oligomers and accompanied change in the exciton delocalization.
- (4) Identification of cyclic and acyclic excitons at single-molecule level through multiparameter analysis.
- (5) Additional (in)coherent excited-state dynamics emerging from energetic and structural manifolds in π -conjugated polymers.

1. Introduction

π -Conjugated polymers (CPs) have played a crucial role in the development of organic photovoltaic devices. Despite their low dielectric constant in organic frameworks, which preferentially generate Frenkel excitons,¹ the charge photogeneration process has proven to be efficient, reaching power conversion efficiencies over 10%. The exact mechanism of charge photogeneration in CPs has remained a controversial issue, but recent studies employing ultrafast spectroscopy have demonstrated that exciton and charge delocalization are a prerequisite for the reduction of electron–hole binding energy.^{1–4} Therefore, understanding the delocalization phenomena in conjunction with the structural properties of polymeric systems is of utmost importance.

CPs are inherently heterogeneous systems. The low torsional barrier between aromatic subunits and the flexibility along the chain allow for numerous isoenergetic conformers containing short π -conjugated segments.⁵ These chain segments couple to each other to create exciton manifolds that are similar to the band structures of semiconductors. Photogenerated high-energy excitons are initially delocalized over the electronically coupled conjugated segments.⁶ This state is coherently funneled to a lower-energy segment, where subsequent structural relaxation and incoherent energy transfer occur.

The complexity of excited-state dynamics can be reduced by employing model oligomeric systems. Ultrafast localization by a coherent energy transfer process and long-range energy migration can be excluded in short oligomers to exclusively investigate the vibration assisted self-trapping and structural relaxation dynamics. The size of the molecule can be systematically controlled to determine the extent of exciton delocalization at various timescales. Shape and conformational disorder can also be controlled to study the exemplary CP structures such as random coil, toroid, and rod. While excited-state dynamics in

^a Department of Chemistry and Spectroscopy Laboratory for Functional π -Electronic Systems, Yonsei University, Seoul 03722, Korea. E-mail: dongho@yonsei.ac.kr

^b Department of Chemistry and Medical Chemistry, Yonsei University, Wonju, Gangwon 26493, Korea

oligomeric systems can be studied in correspondence to the polymeric counterparts, the simplicity and controllability of oligomers facilitate the comprehensive understanding of the nature of CPs.

In this tutorial review, we aim to describe the excited-state structural relaxation dynamics and associated exciton localization and delocalization processes in model π -conjugated oligomers. Emphasis is placed on π -conjugated oligothiophenes, which have been recently studied by our group using fluorescence upconversion spectroscopy to explicitly monitor the evolution of exciton coherence using different spectral parameters. First, we introduce the Hamiltonian approach used to model the energy states and emission properties of linear and cyclic systems. The relationships between the exciton coherence length and fluorescence vibronic peak ratio with and without disorder are outlined. In the following sections, spectroscopic approaches for the characterization of excited-state dynamics of linear and cyclic oligomers are summarized. In the last section, the excited-state dynamics that only exist in polymer systems is

introduced and the significance of exciton delocalization in photovoltaic devices is discussed.

2. Comparison of the electronic structures of linear and cyclic systems

π -Conjugated oligomers share common photophysical characteristics with H/J-aggregates because their constituent subunits are electronically coupled to generate delocalized excitons.⁷ Spano *et al.* and Barford *et al.* have conducted theoretical studies on the absorption and fluorescence properties of π -conjugated oligomers with different structures to show that it is possible to relate the extent of exciton delocalization to the spectral parameters as in H/J-aggregates.^{8–12} In this section, we briefly introduce the basis of the Hamiltonian approach to derive the relationship between the exciton coherence length and vibronic peak ratios in linear and cyclic π -conjugated oligomers. We also discuss the effect of conformational disorder



Kyu Hyung Park

Kyu Hyung Park received his BS (2012) and PhD (2018) degrees from the Department of Chemistry at Yonsei University. He joined Prof. Gregory Scholes's group at Princeton University in 2018 as a postdoctoral fellow.



Woojae Kim

Woojae Kim was born in 1990 in Seoul, Korea. He received his BS degree (2014) from the Department of Chemistry at Yonsei University and now is a PhD student in Prof. Dongho Kim's laboratory.



Jaesung Yang

Jaesung Yang received his BS (2005) and PhD (2011) degrees in Chemistry from Yonsei University. After postdoctoral research with Prof. Dongho Kim at Yonsei University and Prof. Laura J. Kaufman at Columbia University, in 2018, he joined the Department of Chemistry and Medical Chemistry at Yonsei University as an Assistant Professor.



Dongho Kim

Dongho Kim received his BS (1980) from Seoul National University and PhD (1984) from Washington University. After postdoctoral research at Princeton University, he joined the Korea Research Institute of Standards and Science (1986). In 2000, he moved to Yonsei University as a Professor of Chemistry and is now an Underwood Distinguished Professor. He received the Scientist of the Month Award (1999), the Sigma-Aldrich Award (2005), the Korea Science Award in Chemistry (2006), the Star Faculty Award (2006), the National Science Achieving Excellence Award (2010), FILA Basic Science Award (2017), and Academic Excellent Prize in Korean Chemical Society (2018). His research activity has been focused on the excited-state properties and dynamics of various π -conjugated systems.

on limiting the coherence length in these oligomeric systems and describe the concept of spectroscopic unit.

2.1. Influence of geometry in the Frenkel–Holstein model

A Frenkel exciton is a tightly-bound electron–hole pair with a small interchange distance. Analogous to the particle-in-a-box model, the center-of-mass of a Frenkel exciton can be delocalized over a number of monomers that comprise the π -conjugated chains. They couple to the local vibrational modes, typically to C=C stretching modes ($\approx 1400\text{ cm}^{-1}$) that are commonly found in organic chromophores. This coupling can be adequately described by the Frenkel–Holstein model. For a chain composed of N monomers, the Frenkel–Holstein Hamiltonian in the k, q space reads

$$H = \hbar\omega_{0-0} + \hbar\omega_{\text{vib}}\tilde{b}_{q=0}^\dagger\tilde{b}_{q=0} + \hbar\omega_{\text{vib}}\sum_{q\neq 0}b_q^\dagger b_q + \frac{\hbar\omega_{\text{vib}}\lambda}{\sqrt{N}}\sum_{k,q\neq 0}\left\{|k\rangle\langle k+q|b_q^\dagger + |k\rangle\langle k-q|b_q\right\} + \sum_k\tilde{J}_k|k\rangle\langle k| + \left(1 - \frac{1}{N}\right)\hbar\omega_{\text{vib}}\lambda^2, \quad (1)$$

where $\hbar\omega_{0-0}$ is the transition energy of a constituent monomer and $\hbar\omega_{\text{vib}}$ is the energy of the local phonon mode. $|k\rangle$ describes a delocalized exciton with wave vector k , and q is the wave vector of a delocalized phonon mode.^{8,9} Phonons are described by creation and annihilation operators, b_q^\dagger and their Hermitian conjugate, b_q , respectively. $q = 0$ and $q \neq 0$ indicate totally symmetric and non-totally symmetric phonons, respectively, and λ^2 is the Huang-Rhys (HR) factor. \tilde{J}_k is the exciton coupling given by

$$\tilde{J}_k = 2J_{n,n+1}\cos(k) \quad (2)$$

where $J_{n,n+1}$ accounts only for the nearest neighbor exciton coupling with periodic boundary condition. Eigenstates of the Hamiltonian in eqn (1) can be written as

$$|\Psi_{\kappa,\alpha,\tilde{n}_{q=0}}\rangle = |\tilde{n}_{q=0}\rangle \otimes \sum_k \sum_{n_{q_1}, n_{q_2}, \dots, n_{q_{N-1}}} C_{k;n_{q_1}, n_{q_2}, \dots, n_{q_{N-1}}}^{\kappa,\alpha} |k; n_{q_1}, n_{q_2}, \dots, n_{q_{N-1}}\rangle. \quad (3)$$

$\kappa = k + n_{q_1}q_1 + n_{q_2}q_2 + \dots + n_{q_{N-1}}q_{N-1}$ indicates the total momentum and $\tilde{n}_{q=0}$ is the number of totally symmetric phonons, both of which serve as good quantum numbers. Here we assume the temperature $T = 0\text{ K}$ so that κ is not perturbed by thermal energy. α is used to denote the states in order of increasing energy. (Please refer to ref. 7 and 49 for a graphical representation of the energy states and coupling between them.) Emission is stated as transitions between the lowest energy state, $|\Psi_{\kappa=0,\alpha=1,\tilde{n}_{q=0}=0}\rangle$, and the ground state, $|g; n_{q_1}, n_{q_2}, \dots, n_{q_{N-1}}\rangle$, where g indicates the ground electronic state, and following n_{q_l} denotes the vibrational quantum. The intensity of the 0–0 emission band is

$$I^{0-0} = \frac{1}{\mu^2} \left| \langle g; 0, 0, \dots, 0 | \hat{\mu} | \Psi_{\kappa=0,\alpha=1,\tilde{n}_{q=0}=0} \rangle \right|^2 \quad (4)$$

and that of the 0–1 emission band is

$$I^{0-1} = \frac{1}{\mu^2} \sum_l \left| \langle g; n_{q_l} = 1 | \hat{\mu} | \Psi_{\kappa=0,\alpha=1,\tilde{n}_{q=0}=0} \rangle \right|^2. \quad (5)$$

Matrix elements of the transition dipole moment operator $\hat{\mu}$, between the ground electronic state, $|g\rangle$, and exciton state, $|k\rangle$, for a straight chain, are

$$\langle g | \hat{\mu} | k \rangle = \sqrt{N}\mu\delta_{k,0} \quad (6)$$

Then for a bent chain, where the angle between transition dipole moments of monomeric units is defined as θ ,

$$\langle g | \hat{\mu} | k \rangle = \frac{\mu}{\sqrt{N}} \sum_n e^{ikn} \{ \sin(n\theta)i + \cos(n\theta)j \} \quad (7)$$

The ratio of the intensities of the 0–0 and the 0–1 emission bands can be reduced to a simple relationship in straight chains⁸

$$S_R \equiv \frac{I^{0-0}}{I^{0-1}} = \frac{N}{\lambda^2} \quad (8)$$

A similar relationship can also be approximated in bent chains⁹

$$S_R \approx \frac{\csc^2(\theta/2)\sin^2(N\theta/2)}{N\lambda^2}. \quad (9)$$

Since N is the number of constituent monomers, the emission intensity ratio in a straight chain, is proportional to the length of the chain as shown in Fig. 1a (inversely proportional to the HR



Fig. 1 Changes in the fluorescence spectra as a function of (a) the number of subunits N and (b) Huang-Rhys factor λ^2 . Exciton bandwidth is set $W = \omega_0 = 1400\text{ cm}^{-1}$ in (a) and (b). (c) Fluorescence vibronic peak ratio S_R as a function of exciton bandwidth. Adapted from ref. 8 with permission. Copyright 2011 American Chemical Society.



Fig. 2 Change in the 0–0 (top) and 0–1 (middle) fluorescence vibronic peak intensities, and the ratio S_R (bottom) as a function of curvature θ . $N = 40$ and $J_{n,n+1} = -200 \text{ cm}^{-1}$ were used. Adapted from ref. 9 with permission. Copyright 2014 American Chemical Society.

factor and independent of the exciton bandwidth as shown in Fig. 1b and c). However, in a regularly bent chain, S_R decreases as a function of θ and reaches $S_R = 0$ when the geometry of the chain is circular at $\theta = 2\pi/N$. This indicates a complete suppression of the 0–0 emission band (Fig. 2). In the circular chain, S_R as a function of N manifests a deflection point that appears before $N/2$, indicating that the exciton delocalization over half of the circular chain reduces S_R . The S_R values in the two limiting cases have also been evaluated with open boundary conditions by Barford *et al.*, who obtained qualitatively identical results.^{10–12}

Although S_R can be an indicator of the length and curvature of the exciton in uniform disorder-free π -conjugated chains, in most cases, static and dynamic conformational disorders are imposed on the system, for which the thermal effects and site energy disorder are required to be considered.

2.2. The role of disorder in electronic transitions

In the presence of thermal fluctuation or static disorder, the size of the exciton does not correspond to the number of constituent monomers. In many cases, excitons in actual systems are localized to a segment of a chain. The extent of delocalization is defined as coherence length, N_{coh} . As thermal energy contributes to κ , the emission at $T > 0 \text{ K}$ is weighted by the Boltzmann distribution factor⁹

$$P_{\kappa} = \frac{\exp[-\Delta E_{\kappa}/k_{\text{B}}T]}{\sum_{\kappa} \exp[-\Delta E_{\kappa}/k_{\text{B}}T]} \quad (10)$$

where ΔE_{κ} is the energy difference between the exciton of $\kappa \neq 0$ and the band bottom. Incorporation of this prefactor P_{κ} alters

the emission intensity ratio to

$$S_{\text{R}}(T) \approx \frac{1}{N\mu^2\lambda^2} \sqrt{\frac{4\pi\hbar\omega_{\text{c}}}{k_{\text{B}}T}} \sum_{\kappa} e^{-\hbar\omega_{\text{c}}\kappa^2/k_{\text{B}}T} |M_{\kappa}|^2 \quad (11)$$

$$\approx \frac{N_{\text{coh}}(T)}{\lambda^2} c(\theta, T),$$

where $N_{\text{coh}}(T) \approx (4\pi\hbar\omega_{\text{c}}/k_{\text{B}}T)^{1/2}$ is the coherence length approximated in the thermodynamic limit of a straight chain and $c(\theta, T) = (N\mu^2)^{-1} \sum_{\kappa} e^{-\hbar\omega_{\text{c}}\kappa^2/k_{\text{B}}T} |M_{\kappa}|^2$ is the conformation factor, which accounts for the curvature of a chain. This shows that even in the presence of thermal fluctuation, S_{R} can be a good indicator of the size and shape of an exciton.

Torsional disorder is an omnipresent conformational disorder that induces off-diagonal disorder in π -conjugated chains. Theoretically, any amount of disorder can induce the localization of excitons in one-dimensional systems. This has carefully been investigated by Barford *et al.* Unlike the strict dipole–dipole approximation of Spano's approach, Barford incorporates superexchange interactions in the exciton coupling as

$$J_{n,n+1} = J^{\text{DD}} + J^{\text{SE}} \cos^2\phi_n \quad (12)$$

where the first term, J^{DD} , represents the dipole–dipole interaction and the second term including J^{SE} , the superexchange interaction at $\phi_n = 0$, represents the dependence of the π -orbital overlap on a torsional angle between two neighboring units.⁶ Unlike the weak torsional angle dependence in J^{DD} , which is often neglected, J^{SE} is modulated by torsional disorder along the chain. It has been shown that severe torsional disorder, known as conjugation break, can function as a nodal point that accommodates the localized excitons in between.⁵ The chain segment defined by conjugation breaks is considered as a spectroscopic unit, which absorbs/fluoresces as an independent chromophore. The sum of through-space and through-bond coupling in eqn (12) can be used to determine the conjugation cut-off angle, as discussed by van Averbek *et al.*;¹³ different definitions have also been introduced in other works.¹⁴ The size, shape, and arrangement of the spectroscopic units are strongly related to the exciton dynamics on various timescales, such as self-trapping, coherent and incoherent energy transfer, dynamic planarization, and radiative and nonradiative decay to the ground state, which will be detailed in the following sections.

3. Linear π -conjugated oligomers

3.1. Electronic structures of linear π -conjugated oligomers: general description

The electronic transitions of linear π -conjugated oligomers are located throughout the UV-Vis-NIR region where the position of the S_1 transition strongly depends on the monomeric unit (*e.g.*, phenylene, fluorene, thiophene, and porphyrin) and/or linkers (*e.g.*, vinylene, ethynylene, and butadiynylene). Assuming ideally linear geometry without torsional disorder, a transition from the ground state to the first singlet excited (S_1) state corresponds to a one-photon allowed B transition (HOMO \rightarrow LUMO transition,

$^1A_g \rightarrow ^1B_u$ for even-numbered and $^1A_1 \rightarrow ^1B_2$ for odd-numbered oligomers). This transition shows an oligomer-length-dependence, and thus can serve as a good indicator of an effective conjugation length, *i.e.*, the length of light absorbing units or chromophores.¹⁵

Moreover, π -conjugated oligomers have a strong correlation between electronic transitions and vibrational modes: electron-phonon coupling in which two vibrational modes are mainly involved (the high-frequency C=C stretch at $\approx 1400\text{ cm}^{-1}$ and low-frequency inter-ring torsion at $\approx 100\text{ cm}^{-1}$).¹⁶ On account of the inherently strong electron-phonon coupling in π -conjugated systems, the manner in which these vibrational modes and molecular geometries affect the shape of the steady-state linear electronic spectra can be deduced. Typically, at room temperature, the absorption spectra of oligomers are quite broad, whereas the fluorescence spectra show structured vibronic bands that correspond to the break of the mirror image symmetry. These spectral characteristics can be easily understood from the curvatures of torsional potentials, which are closely related to different bond length alternations (BLA) between the ground and S_1 states. On account of the benzoidal bond arrangement (a large BLA, single-bond character between monomeric units) in the ground state, the torsional potential is quite shallow. Therefore, at room temperature ($k_B T \sim 200\text{ cm}^{-1}$) the broad distribution of conformers with nonplanar equilibrium induces an inhomogeneous broadening of the absorption spectra. In contrast, a quinoidal bond arrangement (a small BLA, double-bond character between monomeric units) is formed in the S_1 state; thus, the torsional potential becomes sharper in comparison with the ground-state potential. Upon photoexcitation, the broadly distributed (torsionally disordered) Franck-Condon geometries undergo torsional relaxation along the steep excited-state potential, thus narrowing the torsional distribution near the planar equilibrium. This results in the narrower fluorescence spectra.

Within the Franck-Condon approximation, the HR analysis for narrowed fluorescence spectra allows the measurement of the extent of electron-phonon coupling (HR factor), which is proportional to the square of the displacement between the potential minima of the ground and excited states for a certain vibrational mode.¹⁷ In π -conjugated oligomeric systems, a single vibrational mode, the C=C stretch, is good enough to simulate vibronic progression in the solution-phase fluorescence spectra at room temperature because this vibrational mode is most strongly coupled to the electronic transition between the ground and S_1 states. Furthermore, the HR factor generally decreases as the oligomer length increases, resulting in a relatively intense 0-0 band compared with a small 0-1 side band for longer oligomers. Since π -conjugated oligomers (coupled by through-bond interaction) share similar photo-physical properties with linear J-aggregates (coupled by through-space interaction),⁷ three spectroscopic parameters are relevant for the length of the exciton delocalization along the oligomer chain. These parameters are the peak position of the 0-0 band (E_{0-0}), radiative decay rate (k_r), and fluorescence intensity ratio between the 0-0 and 0-1 bands (I^{0-0}/I^{0-1}). As the oligomer length increases, (1) the 0-0 peak position red-shifts,

(2) the radiative decay rate increases, and (3) the vibronic peak ratio between the 0-0 and 0-1 bands increases.

3.2. Exciton localization processes: the experimental evidence of exciton self-trapping

When conjugated oligomers absorb light, excitons are widely delocalized throughout the oligomer backbone, but within 200 fs, the excitons rapidly localize on conjugated segments of the chain; this is the exciton localization process. This ultrafast relaxation process has been extensively characterized and discussed by several theoretical and experimental approaches in the field of CPs.^{6,18} The following three processes are generally accepted to be important in the exciton localization:¹⁹ (1) electronic relaxation through the slightly delocalized states formed by several coupled chromophoric units, (2) coherent excitation energy transfer (EET) processes among partially delocalized states in the intermediate coupling regime, and (3) self-trapping of delocalized excitons *via* nuclear vibrational modes. Among these, there is still a controversy over which is the most deeply involved in the exciton localization process. However, it has been proposed that the relative contribution of each of the possible processes may vary depending on the type of polymers and the polymer's surrounding environment. For oligomers, the first two processes have little effect on the localization because of their inherently short chain lengths. Therefore, self-trapping through molecular vibrations is the dominant process for the localization of an initially delocalized exciton. Similar to CPs, this ultrafast relaxation process in oligomers is governed by two vibrational modes: C=C stretching ($\sim 25\text{--}30\text{ fs}$ period) and inter-ring torsion ($\sim 160\text{--}250\text{ fs}$ period), which again reflects the strong nature of the electron-phonon coupling in π -conjugated systems.

There is spectroscopic evidence of exciton localization dynamics in conjugated oligomers occurring at the sub-200 fs time scale. The first evidence is the time-resolved fluorescence spectra at early time delay, usually with a 200-300 fs decay that is the earliest measurable spectrum considering the instrumental response function of typical fluorescence upconversion experiments. This can provide indirect but important information on the sub-200 fs relaxation processes. Kim *et al.* reported the reconstructed time-resolved fluorescence spectra of a linear oligothiophene 10-mer (L-10T) in which monomeric thiophenes are linked by ethynylene bridges.²⁰ Certainly, the time-resolved fluorescence spectra at a few hundred femtoseconds are already similar to the steady-state fluorescence spectra and are also narrower than the mirror-image of the absorption spectra. This implies that by the time we observe the fluorescence spectra of the linear oligothiophenes, the exciton localization *via* self-trapping has already finished and the broad distribution of torsionally disordered conformers in the Franck-Condon state is relaxed along the C=C stretch and inter-ring torsional coordinates. Similar spectral behaviors were also observed in time-resolved fluorescence measurements of shorter oligothiophenes with the same ethynylene linkers,²¹ directly-linked oligothiophenes,²² oligofluorenes,²³ and porphyrin oligomers.²⁴

Then the polarization memory loss (anisotropy decay), which can be measured by both transient fluorescence and absorption techniques, provides additional evidence. In these experiments, two parameters may provide a direct insight into the ultrafast exciton localization process *via* self-trapping: the initial anisotropy value and the time constants of ultrafast depolarization. It must be considered that an ideal anisotropy value is theoretically 0.4 when there is no relaxation process in the exciton transition dipole. In particular, the initial anisotropy value also provides indirect but valuable information, although a rapid anisotropy drop from 0.4 cannot be directly observed due to the temporal resolution limit of the spectroscopic apparatus. Kim *et al.* performed this experiment on linear oligothiophenes 4-, 6- and 10-mers (L-4T, L-6T, and L-10T) with ethynylene linkers after photoexcitation at the S_1 state using fluorescence upconversion spectroscopy (Fig. 3).²⁰ The initial anisotropy value of the shortest L-4T is around 0.38, and it does not show any ultrafast depolarization components, implying that the exciton localization process is not so prominent for the short oligomer. As the chain length increases, the initial values gradually decrease (~ 0.36 and ~ 0.28 for L-6T and L-10T, respectively). This suggests an important characteristic regarding molecular structures; in solution, long chains do not exist as a straight structure, but rather in a bent geometry such as a worm-like structure. Therefore, when the initially delocalized exciton is localized through the C=C stretches, the loss of polarization memory should be more prominent in the

longer oligomer.²⁵ Furthermore, for L-6T and L-10T additional ultrafast depolarization components were detected with time constants of ~ 200 and ~ 400 fs, respectively. These were assigned to the self-trapping process through inter-ring torsion, which leads to not only exciton localization but also local planarization. For butadiyne-bridged porphyrin tetra-, hexa-, and octamers, Chang *et al.* reported an oligomer-length-dependent decline of the initial fluorescence anisotropy values.²⁴ The authors synthesized a double-stranded octamer ladder as a reference for the isolated octamer. This ladder-like structure has a rigid backbone and suppressed torsion; they observed a much higher initial anisotropy value (~ 0.34) compared with the worm-like octamer chain (~ 0.24). This is in strong agreement with the fact that the ultrafast depolarization originates from the exciton self-trapping process and is greatly influenced by molecular conformation. Based on transient absorption anisotropy measurements, Dykstra *et al.* studied oligo-*para*-phenylenevinylene 4-, 6-, and 8-mers and observed a slightly lower initial anisotropy value only for the 8-mer.¹⁴ In this study, they suggested that the entropy due to the relatively large conformational disorder of the 8-mer presumably has significant implications for the ultrafast depolarization.

Finally, additional evidence of the exciton localization in conjugated oligomers was experimentally observed by Dykstra *et al.* using three-pulse photon echo peak shift (3PEPS) spectroscopy.²⁶ This tool is known to be useful for probing the early time dynamics hidden by ensemble averaging in linear spectroscopy. The authors

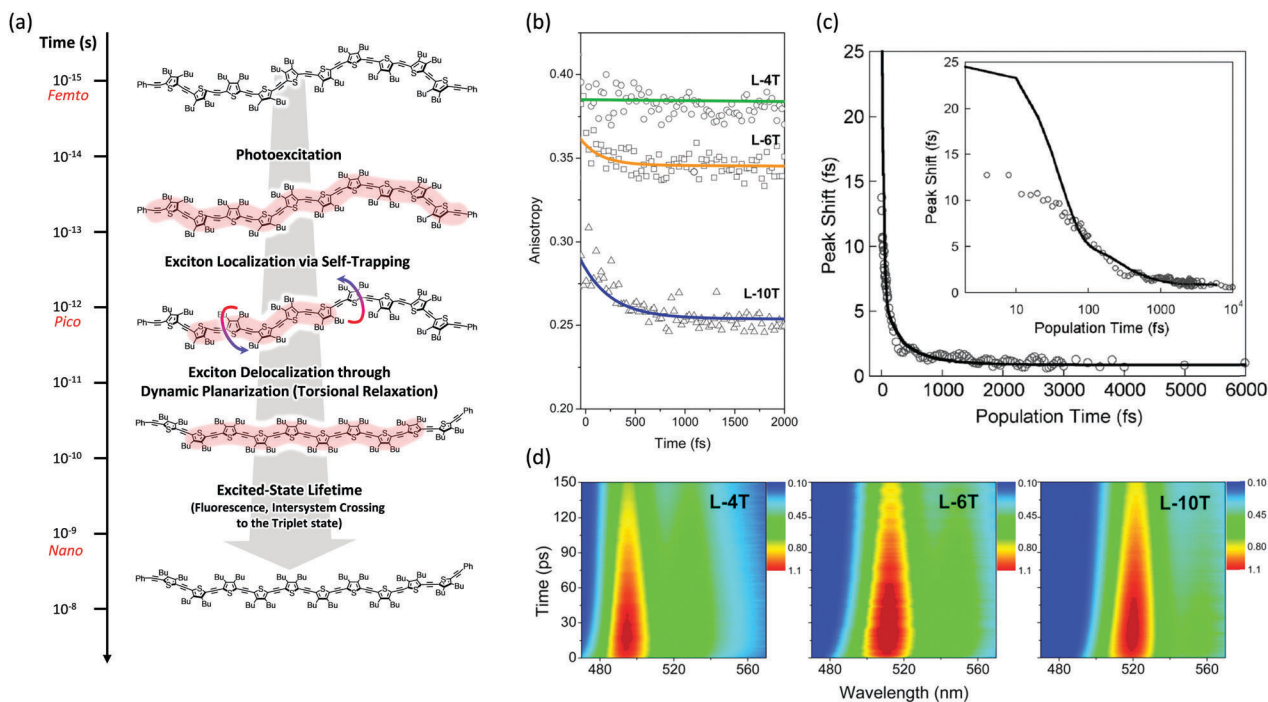


Fig. 3 (a) Schematic representation of excited-state dynamics of linear oligothiophenes from femto- to nanosecond timescales. Image modified from ref. 25 with permission. Copyright 2015 American Chemical Society. (b) Fluorescence anisotropy decays of linear oligothiophene 4- (green), 6- (yellow), and 10-mers (blue). Adapted from ref. 21 with permission. Copyright 2016 American Chemical Society. (c) Three-pulse photon echo peak shift (3PEPS) of the MEH-PPV pentamer (inset: the same data in log-x scale). Adapted from ref. 26 with permission. Copyright 2005 American Physical Society. (d) Transient fluorescence spectra of oligothiophene 4- (left), 6- (middle), and 10-mers (right). Adapted from ref. 21 with permission. Copyright 2016 American Chemical Society.

directly observed a rapid peak-shift decay on a time scale of ~ 25 fs for a pentameric analog of MEH-PPV, which corresponds to the rapid exciton localization *via* self-trapping with the C=C stretch (Fig. 3). Furthermore, as compared to the MEH-PPV polymer, the pentamer shows a lower asymptotic offset as well as longer damped oscillation. These results clearly reflect the lower degree of conformational disorder and structural defects in conjugated oligomers.

3.3. Exciton delocalization processes through dynamic planarization

After the ultrafast localization process of the initially delocalized exciton in linear conjugated oligomers at a sub-ps regime, an additional relaxation process occurs with time constants of a few picoseconds to a few tens of picoseconds. This is known as the dynamic planarization process, which is also called torsional relaxation. Through this process, the torsional angle between the monomeric units becomes small, thereby planarizing the molecular backbone and reducing the inhomogeneous broadening. Moreover, the process induces exciton delocalization along the oligomer chain, which further lowers the excited-state energy. Until the early 1990s, the origin of the intraband relaxation dynamics of oligothiophenes was unclear. In 1996, by measuring the transient stimulated emission using a pump-probe technique, Lanzani *et al.* firstly suggested this dynamics showing a 4 ps time constant as a torsional relaxation process, which involves inter-ring torsional motion.²⁷ Two years later, Wong *et al.* directly measured the dynamic red-shift occurring in a few picoseconds in the time-resolved fluorescence spectra of hexamethylsexithiophene using a fluorescence upconversion technique.²⁸ This experimentally proved that the intraband relaxation process is a redistribution dynamics of excess excitation energy through torsional relaxation that enhances π -electron delocalization in the excited state. Furthermore, by applying coherent vibrational spectroscopy for a substituted quinquethiophene molecule, Cirimi *et al.* directly observed torsional wavepackets (~ 140 cm^{-1}) which consequently lead to the excited-state planarization.²⁹ Chain-length-dependent conformational relaxation dynamics in oligofluorenes was investigated by Hintschich *et al.* By using picosecond single photon counting and a streak camera, they found that in contrast to a polyfluorene derivative, the relaxation in oligofluorenes involves entirely conformational dynamics with time constants ranging between 10 and 300 ps, which decreases with decreasing oligomer length and viscosity of the solvents or increasing temperature.²³ Furthermore, Chang *et al.* suggested that the vibronic peak ratio between the purely electronic 0-0 band and higher vibronic peaks is an indicator of torsional relaxation dynamics.²⁴ This is because of the strong electron-phonon coupling in conjugated oligomers. Therefore, the HR factor (along the C=C stretch coordinate) can be estimated when analysing the vibronic peak ratio from the time-resolved fluorescence spectra. The time-dependent variation of this factor suggests that the exciton delocalization process occurs through dynamic planarization. Moreover, the same authors suggested that the rise of total emission intensity indicates dynamic planarization

because the exciton delocalization causes a growth in the emitting oscillator strength. The similar spectral behaviors were also revealed in oligothiophenes with ethynylene linkers reported by Kim *et al.*^{20,21} They found that the time scale of rise dynamics in total emission intensity profiles is in line with the dynamic red shift and spectral narrowing, proving that indeed all the spectral dynamics occurring on picosecond time scales are related with exciton delocalization through dynamic planarization. In particular, the shorter time constants of planarization processes of oligothiophenes, where the thiophene units are directly linked, were reported by Gallaher *et al.*²² This is due to the result that different force constants (stiffness) of excited-state torsional potential arise from the possession of conjugated linkers.¹⁶

4. Cyclic π -conjugated oligomers

Cyclic geometry has two major differences when compared with its linear counterpart: (1) the symmetry of the system which alters the electronic transition between the ground and emitting states and (2) the reduced degree of freedom, *i.e.*, conformational disorder, of the two termini due to the formation of a closed loop.²⁰ The former has a direct impact on the spectroscopic parameters such as extinction coefficient, fluorescence quantum yield and lifetime, and fluorescence vibronic peak ratios. The latter influences the extent of exciton delocalization, which is a function of torsional disorder, as suggested by Barford *et al.*^{11,12}

4.1. Extent of exciton delocalization in π -conjugated cyclic oligomers

Unlike the optically allowed B transition of the linear π -conjugated systems, the lowest-energy transition of the cyclic counterpart has A symmetry, which is optically forbidden. The characteristics of the electronic transitions of cyclic systems have been extensively investigated with the light harvesting complex (LH) of *Rhodospseudomonas acidophila*, an ideal noncovalently constructed assembly. Despite the fundamental difference in the interchromophore coupling mode, the description of electronic structures of the LH complex is identical to that of π -conjugated cyclic oligomers.³⁰ The impact of disorder parameters, such as site energy disorder, elliptical deformation, and symmetry-breaking gap, on the electronic structures of the LH complex can also be applied to π -conjugated cyclic oligomers with conformational disorder.³¹ Again, this suggests that the electronic structures of CPs can be reasonably described by considering only dipole-dipole coupling and diagonal disorder, as suggested by Spano *et al.*^{8,9}

The electronic structure of a cyclic system comprising n monomeric units comprises $2n$ states, which are rearranged by the nearest neighbor coupling ($V \neq 0$) of two nondegenerate states of each monomer, as shown in Fig. 4. For the cyclic structure with no disorder ($\Delta = 0$), the lowest-energy excited-state, $k = 0$, is optically forbidden. Higher energy states form doubly-degenerate states ($k = \pm 1, \pm 2, \pm 3, \dots$), where only $k = \pm 1$



Fig. 4 Energy diagram of the electronically excited states of cyclic π -conjugated oligomers. Top panel indicates the nondegenerate energy levels of n monomeric units and bottom left panel shows the rearranged energy levels upon nearest-neighbor interaction. Size of the green circles and direction of the double-headed arrow indicate oscillator strength and direction of transition dipole moment of a given state, respectively. Bottom right panel shows the change in the energy levels and oscillator strength upon addition of disorder. Image modified from ref. 30 with permission. Copyright 2001 Cell Press.

states carry oscillator strength. The transition dipole moments of $k = \pm 1$ are in plane and perpendicular to each other. The $k = 0$ state gains a small oscillator strength through Herzberg–Teller (HT) coupling to non-totally symmetric vibrational modes, which dynamically deform the cyclic structure to transiently produce a localized deformation. The effect of HT coupling on electronic transitions in cyclic systems has been carefully investigated by Sprafke *et al.* in a six-porphyrin nanoring, which is rigidly held by a hexapod template molecule.³² Through simulation of steady-state fluorescence spectra, the authors showed that excitons generated in the nanorings weakly fluoresce through HT coupling to a non-totally symmetric vibrational mode, which is the stretching motion of butadiyne linkers at 2244 cm^{-1} . This vibrational mode significantly alters the bond lengths of that part of the rim, developing localized transition densities that resemble those of the optically allowed S_0 – S_2 and S_0 – S_3 transitions.

Aside from this dynamic symmetry-lowering motion, the lowest energy transition is influenced by static disorder. Addition of disorder ($\Delta \neq 0$) lifts the pairwise degeneracy and redistributes the oscillator strength. The $k = 0$ state, which is optically forbidden in the absence of disorder, gains appreciable oscillator strength. The gradual increase in the conformational disorder is evident in size-dependent changes in the steady-state absorption spectra of cyclic oligothiophenes, as investigated by Park *et al.*³³ The lowest energy transition of the smallest cyclic oligothiophene composed of 6 thienylene-ethynylene subunits (C-6T), corresponding to the tailing band from 450 to 550 nm in Fig. 6, is weakly allowed by the HT mechanism. As the size of the cyclic system increases, this band gradually intensifies. For the largest cyclic oligothiophene with 12 subunits (C-12T), the absorption spectrum is structureless, indicating that large conformational heterogeneity similar to

that of linear systems is present. The effect of increasing conformational disorder is also observed in other fluorescence parameters, such as fluorescence quantum yields and fluorescence lifetimes, yet it is most noticeable in the ratio of 0–0 to 0–1 fluorescence vibronic bands. In the smallest cyclic oligothiophene, C-6T, the 0–0 vibronic band is greatly suppressed compared to the 0–1 one, whereas in the largest cyclic oligothiophene, C-12T, the 0–0 is more intense than the 0–1 vibronic band, similar to typical fluorescence spectra of linear systems. The decrease in I^{0-1}/I^{0-0} when the size of the system increases indicates that based on the model by Spano *et al.*, the extent of exciton delocalization is limited. Thus, the size of the exciton relative to the size of the conjugation pathway of cyclic oligothiophenes decreases when the size of the system is increased. Such a trend is also observed for other π -conjugated cyclic oligomers, for example, cycloparaphenylenes,³⁴ porphyrin nanorings,³⁵ and cyclothiophenes,³⁶ but systematic analyses on the extent of exciton delocalization have seldom been reported.

In cycloparaphenylene systems, the size of exciton delocalization was investigated by Adamska *et al.*³⁴ Quantum mechanical calculations of the excited-state geometry including the solvent have shown that cycloparaphenylenes with less than 8 subunits generate fully delocalized excitons, whereas those composed of more than 8 subunits develop localized excitons upon relaxation to the lowest excited-state geometry, S_1' (Fig. 5). Nonadiabatic excited-state dynamics (NA-ESMD) simulations suggest that the self-trapping of excitons in these systems is an ultrafast process that occurs within 50 fs. Exciton localization in porphyrin nanorings has been studied in detail by Parkinson *et al.*³⁵ The initial fluorescence anisotropy value of 0.1 in porphyrin nanorings comprising 20 or less subunits indicates that initially formed excitons are fully delocalized. However, those with more than



Fig. 5 (a) Transition density and (b) transition dipole moment of cycloparaphenylene 6-, 9-, and 12-mers at S_1 , S_2 , S_3 , and relaxed S_1' states. Adapted from ref. 34 with permission. Copyright 2014 American Chemical Society.

20 subunits exhibit initial anisotropy values over 0.1, followed by additional picosecond depolarization. Combined with Monte Carlo simulations, it was shown that in the 40-porphyrin nanoring, both initial and final excitons behave as linear excitons, and this behavior is similar to that in linear porphyrin arrays.

There is a clear dependence between the size of the initial excitons generated upon photoexcitation and the following self-trapping process, as demonstrated in cycloparaphenylenes and porphyrin nanorings. In a picosecond timescale, the slower planarization of the backbone chain re-extends the localized exciton. This has been studied in the transient fluorescence spectra of cyclic oligothiophenes and analyzed in terms of spectral red-shift, total fluorescence intensity (Total I_{FL}), and 0–1 to 0–0 vibronic peak ratio (I^{0-1}/I^{0-0}).³³ As shown in the transient fluorescence spectra (Fig. 6c–f), the cyclic oligothiophenes comprising 6 and 8 subunits (C-6T and C-8T) present no apparent spectral red-shift, but those with 10 and 12 subunits (C-10T and C-12T) display a distinct red-shift. This indicates that the excitons are effectively delocalized in 10- and 12-mers, whereas excitons are already fully delocalized along the whole cyclic framework in 6- and 8-mers. Nonetheless, the total fluorescence intensity abruptly drops in 6- and 8-mers with an exponential decay time constant (amplitude) of 11 (47.3%) and 26 (48.8%) ps, respectively, which coincide with the typical time constants of the dynamic planarization process. Comparable picosecond decay components were found in the total fluorescence decays of 10- and 12-mers with time constants of 49 (33.4%) and 66 (29.5%) ps, respectively. A similar size-dependent trend was observed in the rise dynamics of I^{0-1}/I^{0-0} indicating that this picosecond dynamics is a consequence of the exciton delocalization that regenerates the cyclic symmetry. The large amplitude and short time constant in 6- and 8-mers suggest that dynamic planarization is the process that removes marginal conformational disorder in these systems to attain the ideal cyclic geometry. Then, the large red-shift along with the long time constant and small amplitude of dynamics in 10- and 12-mers suggest that localized excitons are inefficiently delocalized through the dynamic planarization motion, due to the large conformational disorder of the system. Small I^{0-1}/I^{0-0} values of 10- and 12-mers indicate that excitons are not fully delocalized over the whole conjugated backbone and remain as acyclic excitons after the dynamic planarization dynamics.

4.2. Subpopulations of cyclic and acyclic excitons

Fluorescence parameters in ensemble measurements are averaged quantities from various conformers that oligomers can adopt in the given environment. Therefore, the limits of exciton delocalization described in the previous section may not be applicable to the entire population of π -conjugated oligomers when conformational heterogeneity is considered. Single-molecule spectroscopy is the optimal technique to discriminate and identify the behaviors of conformationally heterogeneous systems at the sub-ensemble level. Especially in cyclic systems, where severe conformational disorder such as torsional defects can significantly alter the excitonic nature, a statistical treatment of the fluorescence

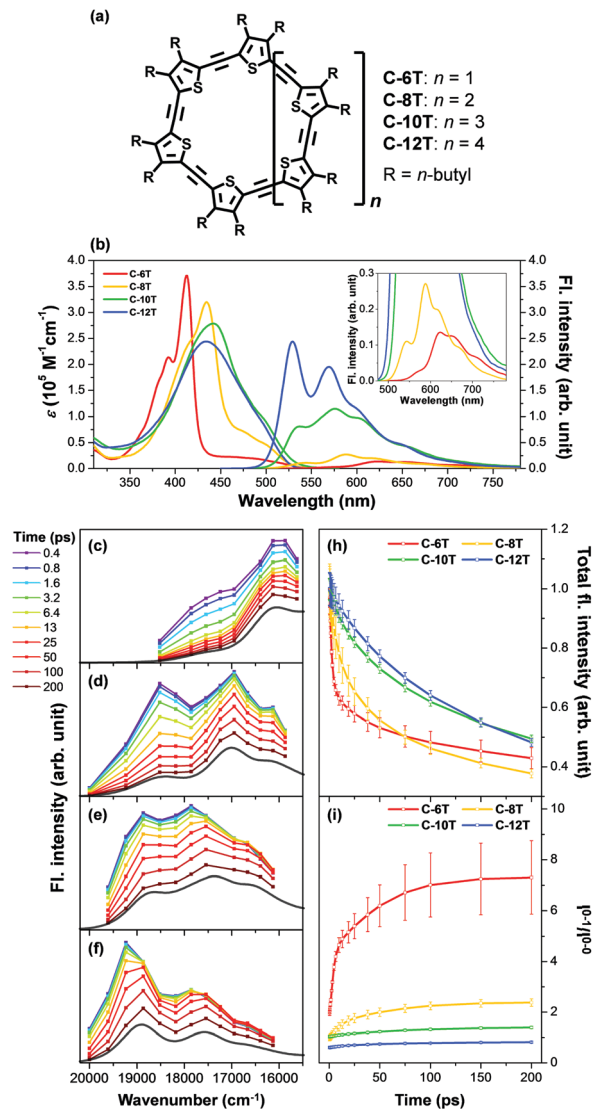


Fig. 6 (a) Molecular structure of cyclic oligothiophenes and their (b) steady-state absorption and fluorescence spectra and (c–f) transient fluorescence spectra. Transient fluorescence spectra are analysed with two to three Gaussian-shaped vibronic peaks and their (h) total fluorescence intensity and (i) 0–1 to 0–0 vibronic peak ratios are plotted against time. Adapted from ref. 33 with permission. Copyright 2015 Wiley-VCH.

parameters can be used to identify the subpopulations elusive in the ensemble measurements.

In an exemplary work by Yang *et al.*, the authors studied a cyclic oligothiophene comprising 10 thienylenes with ethynylene and vinylene linkers (C-10T_{2V}).³⁷ In contrast to the previous ensemble measurements,²⁰ a subpopulation of linear-like excitons at the single-molecule level was additionally found.⁴³ To identify the excitonic features of the molecule, two parameters that have a direct correlation to the structure of the cyclic system, fluorescence lifetime and peak position, were used to construct two-dimensional correlation maps. As shown in Fig. 7a, the initial step displays a single population dispersed along the map. The negative slope of the distribution indicates that fluorescence lifetime and fluorescence peak position are



Fig. 7 Two-dimensional correlation maps between fluorescence lifetime and 0–1 fluorescence vibronic peak position from (a) the initial and (b) subsequent steps before photobleaching. (c) Distribution of peak positions in the first (blue) and subsequent (grey) steps, which correspond to (d) two different excitonic states. (e) Energies and inverse of the square of dipole moments for the transition to the lowest excited state as a function of average torsional angle. (f) Relative ground-state stability as a function of the same parameter is shown to provide information on the thermal distribution. Adapted from ref. 37 with permission. Copyright 2015 American Chemical Society.

anticorrelated parameters, which are modulated by the torsional disorder of the system. In the following step (Fig. 7b), consecutive laser excitation perturbs the initial conformation to induce torsional disorder into the system. This generates a higher energy domain that does not exhibit the linear correlation shown in the initial step.

To account for the different behaviors of the two subpopulations, the lowest-energy transition energy and $1/\mu^2$, which is inversely proportional to the radiative decay rate, are calculated as a function of increasing average torsional angle of the cyclic oligothiophene. As presented in Fig. 7e and f, while the transition energy shows monotonic increase, $1/\mu^2$ displays a deflection point and then remains constant. This deflection point corresponds to the conjugation cut-off angle, as evidenced by the electron density difference map. The generation of a new domain with no transition-energy-responsive fluorescence lifetimes indicates that this subpopulation has a linear character due to the torsional defects generated *via* conformational fluctuations.

Two-dimensional correlation maps can be used when a single fluorescence parameter does not sufficiently support the formation of cyclic excitons. When unambiguous characteristics of cyclic excitons appear, such as long fluorescence lifetime and large I^{0-1}/I^{0-0} , it can be concluded that the system exclusively generates cyclic excitons.³⁸ This is the case for small cyclic oligothiophenes comprising 6 and 8 subunits. Extremely long fluorescence lifetimes (over 2 ns), along with low fluorescence intensity indicate that cyclic excitons with small torsional disorder are generated. Fluorescence lifetimes longer than 1 ns are not found in larger systems with 10 and 12 subunits. Their average fluorescence lifetimes are respectively 590 and 540 ps, which are similar to those of typical linear oligomers. The determination of the extent of the exciton delocalization in the vicinity of the ring-to-linear behavioral turning point again requires two-dimensional correlation maps.

Fig. 8 shows the correlation distribution of fluorescence lifetimes and fluorescence peak positions extracted from the first and the following steps. It is worth noting that new domains at higher energies arise in both the cyclic oligothiophenes. If this

change is associated with the generation of torsional defects in cyclic excitons, according to the discussion provided by Yang *et al.*, the fluorescence lifetime should be significantly decreased. In cyclic oligothiophenes with 10 subunits, the conformational fluctuation modifies the fluorescence lifetime from 620 to 590 ps; this supports that the conformational fluctuations in this system induce a cyclic–acyclic transition of excitons. In cyclic oligothiophenes with 12 subunits, the fluorescence lifetimes remain unaltered upon conformational fluctuation. This suggests that the torsional disorder of this system is positioned beyond the deflection point in Fig. 7e. Excitons generated in this system are initially acyclic due to a large degree of torsional disorder, and subsequent conformation fluctuation only induces a reduction of the delocalization length.

Excitons in much larger systems are not fully delocalized over the whole molecular framework, but their dynamics are still influenced by the geometry of the system. Aggarwal *et al.* studied the exciton dynamics of a giant macrocycle using single-molecule spectroscopy.³⁹ The macrocycle, comprising 6 carbazole-bridged phenylene-ethynylene-butadiynylene subunits, generates localized excitons which span over the size of two subunits. To study the localization dynamics, they measured the excitation and emission linear dichroism, $D_{\text{ex(em)}} = (I_{\text{V}} - I_{\text{H}})/(I_{\text{V}} + I_{\text{H}})$. D_{ex} uses polarized excitation and monitors the emission intensity to determine the anisotropy of the absorption. On account of the rigid cyclic geometry, D_{ex} of the giant macrocycle exhibits a narrow distribution centered at $D_{\text{ex}} = 0$. In comparison to the broad D_{ex} peaks near ± 1 , this indicates that the absorption of the giant macrocycle is ideally isotropic. D_{em} compares the emission intensity of horizontal and vertical polarization to measure the orientation of the emitting state. Again, unlike the broad distribution in the dimer and hexamer, the D_{em} is centered at 0. If the localization dynamics were a deterministic process, which delivers excitons in an identical position for every excitation, the distribution of D_{em} should resemble that of a linear system. This suggests that in a well-defined cyclic system, even acyclic excitons are influenced by the geometry of the cyclic conjugation pathway.



Fig. 8 Two-dimensional correlation maps between fluorescence lifetimes and peak positions of 0–1 vibronic transition of the initial (panel a for 10-mer and panel d for 12-mer) and all the subsequent molecular states (panel b for 10-mer and panel e for 12-mer). The corresponding fluorescence lifetime histogram and spectral distribution are respectively depicted at the top and right side of each panel. (c and f) Schematic illustrations of the change in excitonic states from the first to the next steps. Adapted from ref. 38 with permission. Copyright 2016 American Chemical Society.

4.3. Enhancing exciton delocalization: the effect of π -linkers

Cyclic geometry is certainly advantageous over its linear counterpart to achieve superior exciton delocalization. However, it is not the best approach because π -orbital overlap can be greatly enhanced by chemically restraining the torsional motions.⁴⁰ Such synthetic strategies were proven successful in extending the π -conjugation in polymers. However, they are not applicable in cyclic systems because an increase of ring strain, which is already high in small rings, can hamper the attainment of the ideal structure.

A mild approach that does not impose stress on the structure is the incorporation of rigidifying π -linkers. Liu *et al.* tested single bond, vinylene, and ethynylene linkers between bipyridyl and fluorene moieties in alternating copolymers and found that the ones with vinylene linkers exhibit the most red-shifted absorption and fluorescence spectra.⁴¹ This suggests that vinylene linkers enhance the π -conjugation by planarizing the backbone structure. Kim *et al.* applied this method in cyclic oligothiophenes comprising 12 thienylene-ethynylene subunits, that produce acyclic excitons exclusively.⁴² Upon addition of two and three vinylene linkers, the steady-state absorption and fluorescence spectra are systematically red-shifted. Concomitantly, the I^{0-1}/I^{0-0} ratio increases from 0.781 to 0.915 and radiative rate decreases from 7.23×10^8 to 5.89×10^8 s⁻¹, indicating that the extent of exciton delocalization is enhanced when the number of vinylene linkers is increased.

The nonradiative decay displays a decreasing trend from 14.0×10^8 to 7.81×10^8 s⁻¹ upon increasing the number of vinylene linkers. This suggests that vinylene linkers make the backbone structures rigid, which in turn can affect the dynamic planarization process in the excited state.

To study the exciton delocalization process during dynamic planarization, the transient fluorescence spectra were analyzed. Initial spectral positions at 300 fs of 12-mers with two and three vinylene linkers are red-shifted with respect to those without vinylene linkers, indicating that the Franck–Condon geometry is already planar. Since the exciton size in these systems is clearly larger than half of the ring, the center of mass of the spectrum ($\bar{\nu}_{\text{COM}}$) is more red-shifted in 12-mers with three vinylene linkers. Exponential decays of $\bar{\nu}_{\text{COM}}$ have time constants of 32.4, 25.2, and 25.5 ps for 12-mers without and with two and three vinylene linkers, respectively. Associated indicative I^{0-1}/I^{0-0} ratios have an identical trend of 51.2, 31.4, and 30.7 ps with increasing order of the number of vinylene linkers. Faster dynamic planarization processes for both the fluorescence parameters suggest that vinylene linkers are more rigid bridges and thus reduce the torsional disorder in both the initial Franck–Condon geometry and the final structure obtained after the dynamic planarization process. This demonstrates that the manipulation of π -linkers in strained circular structures can be a promising synthetic strategy to obtain cyclic excitons with superior coherence lengths.

5. Implications on polymeric counterparts

Structurally, a polymer is a chain of interconnected oligomeric duplicates. Thus the main differences between a polymer and its oligomer are the size and additional degree of freedom, which give rise to the long-range inter- and intrachain interactions. The relative strength of inter- and intrachain coupling is modulated by the structure of the polymer chain. In a good solvent, polymers are preferentially in an extended chain structure, whereas in a bad solvent, they collapse into random coils, molten globules, or rod-like structures to minimize the solvent-polymer interactions. In the solid phase, where most optoelectronic applications are conceived, interchain interactions dominate the photoinduced dynamics of the polymer.⁴³ The Coulombic interaction between the electron and hole in exciton or charge transfer states is dramatically reduced by efficient interchain delocalization, and free charge carriers can be generated. We first examine the polymers in solution that adopt stretched conformations. Coherent and incoherent EET processes will be discussed.

5.1. Polymers in solution phase: weak interchain coupling

Unlike a small number of countable monomers in short oligomeric systems, polymers contain numerous subunits that interact to form exciton manifold. In the absence of conformational disorder, their electronic structures can be best described by the band model of semiconductors. However, when pervasive torsional defects or kinks induce inhomogeneous broadening of energy states, as in typical polymers, a sum of interacting chromophores is an adequate description.⁴⁴ On account of numerous energy levels and degrees of inter-chromophore coupling, exciton dynamics in the energy landscape of CPs inevitably includes EET processes.

In an ultrafast timescale (<200 fs), the initial photoexcitation undergoes a coherent EET process. The coherent intrachain EET process has been studied by Collini *et al.* in poly[2-methoxy,5-(2'-ethyl-hexoxy)-1,4-phenylenevinylene] (MEH-PPV).⁴⁵ In CPs, this process occurs in the donor and acceptor segments quantum mechanically entangled by intermediate coupling strength ($U \sim \gamma$, where U is the resonance coupling between the donor and acceptor and γ is the dephasing strength).⁴⁶ Coherence between the energy donor and acceptor chromophores and its phase information are preserved before dephasing. Since EET between the randomly oriented spectroscopic subunits is accompanied by a loss of polarization memory, anisotropy can be a good indicator. By monitoring the time evolution of anisotropy of EET dynamics as a function of coherence time (τ) and population time (T), known as the two-time anisotropy decay (TTAD), it is possible to identify the coherent EET process.

Fig. 9A and B are the TTAD maps of the MEH-PPV nanoparticle in water and MEH-PPV in chloroform. Fig. 9C is the simulated map assuming the empirically obtained finite coherence and population times. The main difference is the TTAD decay along the τ axis. MEH-PPV in chloroform displays



Fig. 9 Two-time anisotropy decay plots for (A) MEH-PPV nanoparticles in water and (B) MEH-PPV in chloroform. (C) Two-time anisotropy decay plots simulated by a multi-exponential model based on experimentally obtained time constants. (D) Initial anisotropy values as a function of coherence time τ for rhodamine 6G (green), MEH-PPV nanoparticles in water (orange), and MEH-PPV in chloroform (blue). (E) Three-dimensional plot of the diagonal amplitude as a function of diagonal frequency and population time T . (F) Peak amplitude of diagonal line (black) and the ratio between the diagonal and antidiagonal widths of the peak at $1/e$ height (red). Adapted from ref. 45 with permission. Copyright 2009 American Association for the Advancement of Science.

τ -dependent variation in the anisotropy, indicating that the transition dipole moment coherently changes direction. In nanoparticles, however, the τ -dependence is negligible as in rhodamine 6G, where EET is unviable (Fig. 9D). Coherent EET dynamics of MEH-PPV was reconfirmed by two-dimensional photon echo (2DPE) spectroscopy. In 2DPE, the two-dimensional electronic spectra along the coherence frequency, ω_τ and rephasing frequency, ω_t are measured at a population time T . Spectra along the diagonal line in the 2DPE map, which is related to the linear absorption spectrum of MEH-PPV, oscillate along T , as shown in Fig. 9E. The peak amplitude and diagonal-to-antidiagonal peak width ratio were analyzed to show the anticorrelation, which is manifested in electronic coherence, between these parameters. This electronic coherence persists for more than 250 fs, which is surprisingly long considering the ambient conditions for the measurements. This means that the long-lived coherence can significantly enhance the EET process of CPs.

Incoherent EET is a process occurring between chromophores that are spatially separated over 5 \AA , and are weakly coupled ($U \ll \gamma$).⁴⁴ The process in this regime is best described by Förster energy transfer (FET), or resonance energy transfer

(RET). The successive FET processes that deliver excitons to lower or isoenergetic states is known as exciton diffusion. Since the directionality of EET can stochastically change depending on mutual orientation and coupling between chromophores in the polymer chain, one-dimensional random walk models are often employed to evaluate the exciton diffusion constants. In poly(3-hexylthiophene) (P3HT), the intrachain exciton diffusion constant was calculated by Healy *et al.* using P3HT chains of controlled lengths.⁴⁷ Two chain ends were functionalized with C₆₀ fullerenes and quenching efficiencies were analyzed with one-dimensional random walk exciton diffusion and RET-mediated quenching. The exciton diffusion coefficient and length were $9.8 \times 10^{-4} \text{ cm}^2 \text{ s}^{-1}$ and 7.0 nm, respectively. These values are comparable to those of C₆₀-functionalized oligofluorenes and are also similar to those reported for P3HT films, where exciton diffusion takes place in the chain-stacked aggregated domains. This implies that the incoherent EET process in P3HT, as well as in other π -conjugated polymer analogues, allows for long-range transportation of excitons to the heterojunctions in photovoltaic materials where the charge transfer reaction takes place, as discussed in the following section.

5.2 Polymers in solid phase: strong intermolecular coupling

CP films are semicrystalline, indicating that their morphologies can be divided into two parts: amorphous and crystalline domains. The former is composed of weakly interacting polymer chains with large conformational disorder. Their structures can be regarded as direct projections of the heterogeneous conformations in the solution phase.⁴⁸ The crystalline domain is composed of cofacial π -stacks of polymer chains. Chain aggregates have greatly reduced conformational heterogeneity because intermolecular interactions between π -conjugated segments and between pendent alkyl chains greatly suppress the torsional degree of freedom in a polymer chain. Typical interchain distances in the π -stacking direction lie in a narrow range of 3.3–3.6 Å, where interchain electronic coupling becomes viable. The features of interchain vibronic coupling in the absorption and fluorescence spectra have been modeled by Spano *et al.* by employing a Hamiltonian approach similar to the one outlined in Section 2.⁴⁹ In both the absorption and fluorescence spectra, suppressed 0–0 vibronic transitions indicate interchain coupling. The relative strength of 0–0 to 0–1 vibronic transitions in the absorption spectra is associated with the strength of interchain coupling J , and that in the fluorescence spectra is associated with interchain exciton delocalization.

Recent studies have demonstrated that charge photogeneration in organic photovoltaic devices employs interchain exciton and charge delocalization in the crystalline domains. On account of the low dielectric constant in organic systems ($\epsilon_r \approx 2$ –4), the exciton binding energy in CPs is as large as ~ 1 eV, which cannot be compensated by thermal energy at room temperature.¹ However, when the electron and hole of a charge transfer exciton are delocalized, the separation between the two charge centers effectively reduces the Coulombic interaction between them and the free charges can be readily

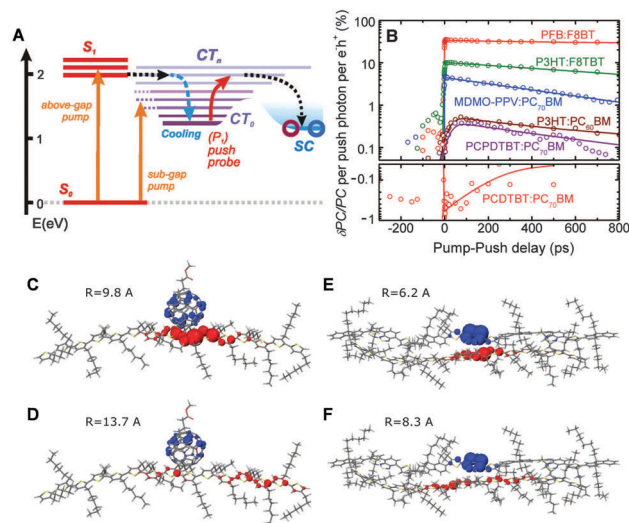


Fig. 10 (A) Energy level diagram of the typical organic photovoltaic (OPV) system. (CT₀, lowest-lying charge transfer state; CT_n, higher-lying charge transfer states; SC, separated-charges states). (B) Time-resolved pump-push photocurrent data for a series of OPVs after photoexcitation at the higher singlet excited state. Computational results of the charge distribution at (C and D) the P3HT/PCBM and (E and F) P3HT/F8TBT heterojunctions. Blue and red lobes indicate electron and hole densities, respectively. (C and E) Simulated charge distribution in the lowest CT-state configuration, and (D and F) in the higher CT state configuration after irradiation of IR-push pulse. Adapted from ref. 2 with permission. Copyright 2012 American Association for the Advancement of Science.

generated. Similar to the coherent EET process, which benefits from the coherence superposition state between the energy donor and acceptor moieties, charge separation processes at heterojunctions benefit from coherent dynamics in an ultrafast timescale. Bakulin *et al.* employed pump-push spectroscopy to access hot charge-delocalized states in various CPs (Fig. 10A).² Fig. 10B shows the differences in photocurrent obtained after a pump pulse above the bandgap energy and the subsequent push pulse in the IR region. All the polymer-based heterojunction materials exhibited an increase in photocurrent except for the PCDTBT:PC₇₀BM blend, which displayed a slight decrease possibly resulting from bimolecular recombination. The increase in photocurrent was monitored using a sub-band gap pump and optical probe. The charge density distributions in higher charge transfer states of molecular heterojunctions were simulated by quantum mechanical calculations, which revealed a drastic increase in the distance between charge centers associated with intrachain charge delocalization (Fig. 10C–F).

The same technique was utilized by Gélinas and Jakowetz *et al.* to measure the electron–hole distance in the charge separation process.^{3,4} The studies focused on the microscopic Stark effect arising from the electric field between loosely bound electron–hole pairs, which cause a red-shift in the absorption spectra, *i.e.*, electroabsorption (Fig. 11B). Gélinas *et al.* prepared blend films with different donor, p-DTS(FBTh₂)₂, and acceptor, PC₇₁BM, ratios. For the 60:40 blend ratio, both molecules form a crystalline phase, whereas at 90:10, the acceptor is intercalated between the donor molecules.³ Fig. 11C presents



Fig. 11 (A) Schematic illustration of photogenerated species at a donor/acceptor interface (1a, excitons in the bulk; 1b, excitons at the interface; 2, electron–hole pairs generating a dipole-like electric field; 3, free charge carriers). (B) Stark shift of absorption spectrum due to an electric field, $S(E)$, and electro-absorption characteristic, EA. (C) Transient absorption spectra of the 100 : 0, 90 : 10, and 60 : 40 films normalized at 640 nm. The steady-state EA spectra are shown for comparison. (D) Peak positions of integrated EA signals of 4 : 1 (blue) and 1 : 4 (red) PCDTBT/mPCBM blends as a function of pump–push delays. Adapted from ref. 3 and 4 with permission. Copyright 2014 American Association for the Advancement of Science and 2017 Nature Publishing Group.

the pump–probe spectra of films with different blend ratios; the steady-state electroabsorption spectrum is shown for comparison. While the 90 : 10 blend film displays only a reduced stimulated emission, the 60 : 40 blend film manifests a clear negative band near 700 nm that matches the position of electroabsorption of the donor molecule. Similar results were obtained using a PCDTBT polymer donor, suggesting that the formation of delocalized states is a prerequisite for an efficient charge separation process. Jakowetz *et al.* investigated this further by utilizing push pulse as an energy input for electron–hole separation.⁴ Pump–probe spectra were measured with and without push pulse, which were subtracted to yield only the signal from electroabsorption. They compared the PCDTBT donor and PC₇₁BM acceptor at 4 : 1 and 1 : 4 ratios, which are respectively characterized by crystalline and disordered phases. The positions of the electroabsorption indicative of the electron–hole separation were plotted as a function of time (Fig. 11D). In the 1 : 4 blend film, charge separation is an energetically uphill process, unfavorable for

long-range charge separation. For the 4 : 1 blend film, the electron and hole move to highly ordered lower-energy sites within 150 fs. Since this ultrafast coherent process delocalizes charges over large distances, estimated at 5 nm, charges can be easily separated afterwards.

The role of exciton and charge delocalization in the charge separation process has been extensively studied using a number of different techniques and is presented in a number of excellent review articles.^{1,50} The oligomer approach in this review shows that exciton dynamics, especially exciton (de)localization processes, can ideally be described in controlled model systems. Such ideality is partially blurred out in the polymeric part; nonetheless, at the microscopic level identical excited-state dynamics are still attained. Concerning this, we found that a careful control and examination of oligomeric model systems will further broaden and deepen the knowledge on complex excited-state dynamics of CPs, which have great potential in organic optoelectronics applications.

6. Conclusions

In this review, excited-state dynamics of π -conjugated oligomers have been illustrated. Special focus was put on the oligothiophenes, which can be considered as model systems for polythiophenes exploited in modern organic photovoltaic devices. We took a bottom-up approach to understand the complex photophysics of CPs. First, general description on the electronic structures and associated absorption and fluorescence characteristics of oligomers was introduced using the Hamiltonian approach by Spano and Barford *et al.* Important relationship between the extent of exciton delocalization and the vibronic peak intensity ratio in the fluorescence spectra was derived. Extent of exciton delocalization was described in chronological order from exciton self-trapping, dynamic planarization, and to the relaxation to the ground state, which occur in femto-, pico-, and nanosecond timescales, respectively. These exciton dynamics were studied in both linear and cyclic oligothiophenes at both ensemble and single-molecule levels. Other analogous oligomeric systems such as oligoparaphenylenes, porphyrin arrays, and oligo(phenylene vinylene) have also been introduced. Finally, excited-state dynamics unique in CPs was outlined in the solution and solid phases. In the solution phase, where interchain interaction is weak, exciton migration processes by coherent and incoherent processes were introduced. In the solid phase, the role of interchain exciton and charge delocalization in the charge generation process was illustrated with recent works employing ultrafast time-resolved spectroscopy. Theoretical and experimental aspects of π -conjugated oligomer and polymer systems reviewed herein will provide a general overview on the excited-state dynamics of these systems crucial in understanding the charge photogeneration processes in organic photovoltaic materials currently studied in related fields.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

First, we thank the colleagues with whom we have been collaborating on this subject, in particular Prof. Masahiko Iyoda (Tokyo Metropolitan University) and his research fellows. The work at Yonsei University was supported by the National Research Foundation of Korea (NRF) grant funded by the Korea government (MEST) (2016R1E1A1A01943379).

Notes and references

- 1 T. M. Clarke and J. R. Durrant, *Chem. Rev.*, 2010, **110**, 6736–6767.
- 2 A. A. Bakulin, A. Rao, V. G. Pavelyev, P. H. M. van Loosdrecht, M. S. Pshenichnikov, D. Niedzialek, J. Cornil, D. Beljonne and R. H. Friend, *Science*, 2012, **335**, 1340–1344.
- 3 S. Gélinas, A. Rao, A. Kumar, S. L. Smith, A. W. Chin, J. Clark, T. S. van der Poll, G. C. Bazan and R. H. Friend, *Science*, 2014, **343**, 512–516.
- 4 A. C. Jakowetz, M. L. Böhm, A. Sadhanala, S. Huettner, A. Rao and R. H. Friend, *Nat. Mater.*, 2017, **16**, 551–557.
- 5 W. Barford, D. G. Lidzey, D. V. Makhov and A. J. H. Meijer, *J. Chem. Phys.*, 2010, **133**, 044504.
- 6 O. R. Tozer and W. Barford, *J. Phys. Chem. A*, 2012, **116**, 10310–10318.
- 7 H. Yamagata and F. C. Spano, *J. Phys. Chem. Lett.*, 2014, **5**, 622–632.
- 8 F. C. Spano and H. Yamagata, *J. Phys. Chem. B*, 2011, **115**, 5133–5143.
- 9 N. J. Hestand and F. C. Spano, *J. Phys. Chem. B*, 2014, **118**, 8352–8363.
- 10 W. Barford and M. Marcus, *J. Chem. Phys.*, 2014, **141**, 164101.
- 11 W. Barford and M. Marcus, *J. Chem. Phys.*, 2016, **145**, 124111.
- 12 M. Marcus, J. Coonjobeeharry and W. Barford, *J. Chem. Phys.*, 2016, **144**, 154102.
- 13 B. Van Averbeke and D. Beljonne, *J. Phys. Chem. A*, 2009, **113**, 2677–2682.
- 14 T. E. Dykstra, E. Hennebicq, D. Beljonne, J. Gierschner, G. Claudio, E. R. Bittner, J. Knoester and G. D. Scholes, *J. Phys. Chem. B*, 2009, **113**, 656–667.
- 15 H. Meier, *Angew. Chem., Int. Ed.*, 2005, **44**, 2482–2506.
- 16 S. Karabunarliev, *J. Phys. Chem. A*, 2000, **104**, 8236–8243.
- 17 A. Yang, M. Kuroda, Y. Shiraishi and T. Kobayashi, *J. Phys. Chem. B*, 1998, **102**, 3706–3711.
- 18 N. Banerji, S. Cowan, E. Vauthey and A. J. Heeger, *J. Phys. Chem. C*, 2011, **115**, 9726–9739.
- 19 N. Banerji, *J. Mater. Chem. C*, 2013, **1**, 3052–3066.
- 20 P. Kim, K. H. Park, W. Kim, T. Tamachi, M. Iyoda and D. Kim, *J. Phys. Chem. Lett.*, 2015, **6**, 451–456.
- 21 T.-W. Kim, W. Kim, K. H. Park, P. Kim, J.-W. Cho, H. Shimizu, M. Iyoda and D. Kim, *J. Phys. Chem. Lett.*, 2016, **7**, 452–458.
- 22 J. K. Gallaher, K. Chen, G. S. Huff, S. K. K. Prasad, K. C. Gordon and J. M. Hodgkiss, *J. Phys. Chem. Lett.*, 2016, **7**, 3307–3312.
- 23 S. I. Hintschich, F. B. Dias and A. P. Monkman, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2006, **74**, 045210.
- 24 M. H. Chang, M. Hoffmann, H. L. Anderson and L. M. Herz, *J. Am. Chem. Soc.*, 2008, **130**, 10171–10178.
- 25 A. Thiessen, D. Würsch, S.-S. Jester, A. V. Aggarwal, S. Bange, J. Vogelsang, S. Höger and J. M. Lupton, *J. Phys. Chem. B*, 2015, **119**, 9949–9958.
- 26 X. Yang, T. E. Dykstra and G. D. Scholes, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **71**, 045203.
- 27 G. Lanzani, M. Nisoli, S. De Silvestri and R. Tubino, *Chem. Phys. Lett.*, 1996, **251**, 339–345.
- 28 K. S. Wong, H. Wang and G. Lanzani, *Chem. Phys. Lett.*, 1998, **288**, 59–64.
- 29 G. Cirmi, D. Brida, A. Gambetta, M. Piacenza, F. Della Sala, L. Favaretto, G. Cerullo and G. Lanzani, *Phys. Chem. Chem. Phys.*, 2010, **12**, 7917–7923.
- 30 M. Matsushita, M. Ketelaars, A. M. van Oijen, J. Köhler, T. J. Aartsma and J. Schmidt, *Biophys. J.*, 2001, **80**, 1604–1614.
- 31 M. F. Richter, J. Baier, T. Prem, S. Oellerich, F. Francia, G. Venturoli, D. Oesterhelt, J. Southall, R. J. Cogdell and J. Köhler, *Proc. Natl. Acad. Sci. U. S. A.*, 2007, **104**, 6661–6665.
- 32 J. K. Sprafke, D. V. Kondratuk, M. Wykes, A. L. Thompson, M. Hoffmann, R. Drevinskas, W.-H. Chen, C. K. Yong, J. Kärnbratt, J. E. Bullock, M. Malfois, M. R. Wasielewski, B. Albinsson, L. M. Herz, D. Zigmantas, D. Beljonne and H. L. Anderson, *J. Am. Chem. Soc.*, 2011, **133**, 17262–17273.
- 33 K. H. Park, P. Kim, W. Kim, H. Shimizu, M. Han, E. Sim, M. Iyoda and D. Kim, *Angew. Chem., Int. Ed.*, 2015, **54**, 12711–12715.
- 34 L. Adamska, I. Nayyar, H. Chen, A. K. Swan, N. Oldani, S. Fernandez-Alberti, M. R. Golder, R. Jasti, S. K. Doorn and S. Tretiak, *Nano Lett.*, 2014, **14**, 6539–6546.
- 35 P. Parkinson, D. V. Kondratuk, C. Menelaou, J. Q. Gong, H. L. Anderson and L. M. Herz, *J. Phys. Chem. Lett.*, 2014, **5**, 4356–4361.
- 36 E. Mena-Osteritz, F. Zhang, G. Götz, P. Reineker and P. Bäuerle, *Beilstein J. Nanotechnol.*, 2011, **2**, 720–726.
- 37 J. Yang, S. Ham, T.-W. Kim, K. H. Park, K. Nakao, H. Shimizu, M. Iyoda and D. Kim, *J. Phys. Chem. B*, 2015, **119**, 4116–4126.
- 38 K. H. Park, J.-W. Cho, T.-W. Kim, H. Shimizu, K. Nakao, M. Iyoda and D. Kim, *J. Phys. Chem. Lett.*, 2016, **7**, 1260–1266.
- 39 A. V. Aggarwal, A. Thiessen, A. Idelson, D. Kalle, D. Würsch, T. Stangl, F. Steiner, S.-S. Jester, J. Vogelsang, S. Höger and J. M. Lupton, *Nat. Chem.*, 2013, **5**, 964–970.
- 40 T. Lei, X. Xia, J.-Y. Wang, C.-J. Liu and J. Pei, *J. Am. Chem. Soc.*, 2014, **136**, 2135–2141.
- 41 B. Liu, W.-L. Yu, J. Pei, S.-Y. Liu, Y.-H. Lai and W. Huang, *Macromolecules*, 2001, **34**, 7932–7940.
- 42 W. Kim, J. Sung, K. H. Park, H. Shimizu, M. Imamura, M. Han, E. Sim, M. Iyoda and D. Kim, *J. Phys. Chem. Lett.*, 2015, **6**, 4444–4450.
- 43 J. Vogelsang, T. Adachi, J. Bazard, D. A. Vanden Bout and P. F. Barbara, *Nat. Mater.*, 2011, **10**, 942–946.

- 44 I. Hwang and G. D. Scholes, *Chem. Mater.*, 2011, **23**, 610–620.
- 45 E. Collini and G. D. Scholes, *Science*, 2009, **323**, 369–373.
- 46 E. Collini and G. D. Scholes, *J. Phys. Chem. A*, 2009, **113**, 4223–4241.
- 47 A. T. Healy, B. W. Boudouris, C. D. Frisbie, M. A. Hillmyer and D. A. Blank, *J. Phys. Chem. Lett.*, 2013, **4**, 3445–3449.
- 48 H. Sirringhaus, P. J. Brown, R. H. Friend, M. M. Nielsen, K. Bechgaard, B. M. W. Langeveld-Voss, A. J. H. Spiering, R. A. J. Janssen, E. W. Meijer, P. Herwig and D. M. de Leeuw, *Nature*, 1999, **401**, 685–688.
- 49 F. C. Spano and C. Silva, *Annu. Rev. Phys. Chem.*, 2014, **65**, 477–500.
- 50 O. Ostroverkhova, *Chem. Rev.*, 2016, **116**, 13279–13412.