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# Coupling free energy and surface anchoring mechanism in gold nanorod–nematic liquid crystal dispersions†

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The self-organizing properties of nematic liquid crystals (NLC) have been widely used to align anisotropic nanorods (NR) dispersed in them. The coupling interaction of the NR–NLC is an important parameter to determine the surface anchoring strength of GNRs in NLC. Herein, we have taken gold nanorods (GNRs) as a typical example to develop a method to measure the coupling free energy between the NRs and NLCs. By probing the magnetic Fréedericksz transition (MFT) thresholds of the doped-LC cells, the extraordinary increasing trend of MFT thresholds in the GNR–LC system can be explained by theory of the Landau–de Gennes free energy. The coupling coefficient  $\gamma$  of the binary composition and the anchoring energy  $W$  of the GNR surface were obtained by choosing proper limitations. Similarly, the MFT thresholds of gold nanospheres (GNSs) and other surfactant-capped GNR were measured for reference and comparison. In addition, response time measurements of the LC cells were carried out to prove the accuracy of the MFT thresholds. These results provide insight into the GNR–LC binary systems, and the technique can be extended to more NR–LC compositions.

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## I. Introduction

Realization of self-assembly for well-organized nanoparticles is always the core target in nanoscience.<sup>1–4</sup> As anisotropic materials, nanorods need a specific and suitable medium to maintain their order and orientation.<sup>5,6</sup> Liquid crystals (LCs), owing to their large anisotropy, shape-proximity to nanorods, and extreme sensitivity to small external perturbations, are a good surrounding to align the nanorods.<sup>7–10</sup>

In the past few years, the gold nanorod (GNR)–liquid crystal (LC) composition as a typical case of NR–LC binary system has been studied from different perspectives.<sup>11,12</sup> GNR can well-disperse and can be long-range ordered in nematic LC with no aggregation.<sup>13</sup> The mixture can be realigned, just like pure LCs, under shearing and electric fields. Moreover, by choosing different aspect ratios of GNR or surfactants around the rods, NLC can be anchored in a parallel or homeotropic fashion on the surface of the nanorods;<sup>14,15</sup> thus, the field-assisted alignment of GNR can be induced in selected NLCs. By utilizing the surface plasmon resonance of GNR, whose longitudinal SPR can be tuned in both the visible and infrared spectral range,

electrically tunable plasmonic filters and polarizers can be fabricated.<sup>16,17</sup> However, the mechanism for the alignment of GNR in NLC has not been examined from a free energy perspective, and the interaction between the GNR and NLC has not been studied.

In this study, we further explored the GNR–NLC system and focused on the free energy of the system and the coupling effect between the GNR and NLC.<sup>18</sup> The interaction theory for ordered dispersion of an anisotropic rod-like nano-material in nematic hosts is applied in our case by giving Landau–de Gennes description of free energy.<sup>19</sup> It is found that for higher concentrations of GNRs, the main change in free energy of the system is dominated by the coupling free energy term  $f_{\text{GNR/NLC}}$ . Owing to this additional free energy, an extraordinary increase in the MFT threshold of the LC cell was observed as expected. By combining the GNR–NLC free energy model and the experimental result of the MFT threshold, both the coupling coefficient  $\gamma$  and the interface anchoring energy  $W$  were obtained.<sup>20</sup> Gold nanospheres or GNRs with different surfactants were doped into a 5CB LC cell for comparison and reference.<sup>21</sup> Time-resolved measurements of the GNR–NLC were carried out to verify the MFT threshold.

## II. Theoretical background

### A. Anchoring effect for a single GNR

On the border of a single NR surface, the director of the LC will tend to be aligned along a certain direction due to the anchoring effect. Upon highly doping GNRs into NLCs, it is

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necessary to take the anchoring energy of the NLC molecules on the GNR surface into account. In this case, the anchoring effect will obey a Rapini–Papoular type of expression for the surface energy.<sup>22,23</sup> By distinguishing the polar anchoring from the azimuthal anchoring, the surface energy  $W$  per unit area can be written as follows:

$$W = W_{\text{iso}} + W_{\text{polar}} + W_{\text{azi}} = W_{\text{iso}} + W_{\alpha} (\vec{p} \cdot \vec{n}) + W_{\beta} (\vec{q} \cdot \vec{n}) \quad (1)$$

where  $W_{\text{iso}}$ ,  $W_{\text{polar}}$ , and  $W_{\text{azi}}$  are the isotropic, polar, and azimuthal anchoring energies, respectively, and  $\vec{p}$ ,  $\vec{q}$ , and  $\vec{n}$  stand for the GNR main axis, GNR side surface normal, and director of the NLC, respectively. If  $W_{\text{polar}} > W_{\text{azi}}$ , the NLC favors an alignment along the GNR main axis; for  $W_{\text{polar}} < W_{\text{azi}}$ , the NLC preferentially aligns perpendicular to the surface of the rods. Both the perpendicular alignment to the surface of the rods or along the GNR main axis can be realized by surface modification with DMOAP (*N,N*-dimethyl-*N*-octadecyl-3-aminopropyl trimethoxysilyl-chloride) or PEG-SH (thiol-terminated methoxy-poly(ethylene glycol)),<sup>13,14</sup> and the anchoring energies are enslaved by different surface modifiers (Fig. 1).

## B. Pure LC cell with limited boundary under an external magnetic field

In all the experiments hereinafter, the LC cell is subjected to an external magnetic field  $H$ , which is parallel to the substrate, to induce the MFT into the LC. Thus, the total free energy function of pure LC cells under an external magnetic field per area on the substrate can be written as follows:<sup>3,22</sup>

$$f = \int_0^d \left[ (K_{11} \sin^2 \theta + K_{33} \cos^2 \theta) \left( \frac{d\theta}{dz} \right)^2 - \chi_a H^2 \sin^2 \theta \right] dz + \frac{1}{2} [\delta(z) + \delta(dz)] \sum_{n=1}^{\infty} W_{2n} \sin^{2n} \theta \quad (2)$$

$K_{11}$  and  $K_{33}$  are the splay and bend elastic constants, respectively,  $\chi_a$  is the magnetic anisotropy of the NLC, the LC cell is confined between  $z = 0$  and  $d$ ,  $\theta$  is the angle between the LC director and the  $z$ -axis, and  $W_{2n}$  with a positive integer  $n$  describes the surface anchoring energy of the LC on the substrates. We can assume that  $W_{2n} = 0$  for  $n > 2$  for simplicity.

Using the variational method, we can obtain the Euler equation for the system in an equilibrium state:

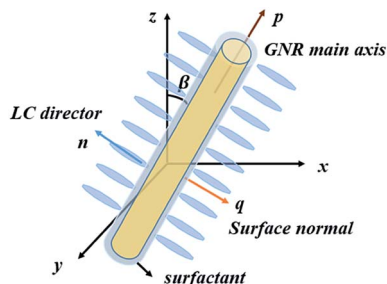


Fig. 1 Schematic of the nematic LCs perpendicularly aligned to the surface of the rod.

$$(K_{11} \sin^2 \theta + K_{33} \cos^2 \theta) \frac{d^2 \theta}{dz^2} - (K_{33} - K_{11}) \cos \theta \sin \theta \left( \frac{d\theta}{dz} \right)^2 - 2\chi_a H^2 \sin \theta \cos \theta = 0 \quad (3)$$

Under boundary conditions,

$$(K_{11} \sin^2 \theta + K_{33} \cos^2 \theta) \frac{d\theta}{dz} \Big|_{z=0,d} = -2W_2 \sin \theta_0 \cos \theta_0 \quad (4)$$

$\theta_0$  is the orientation of the director of the LC as the substrates deviates from the easy direction, which provides resistance to hold back the tendency of the LC to align in the magnetic field direction. Herein,  $\theta = \theta(z)$  implies that under a magnetic field, LC at different  $z$ -axis positions will orient with different angles, and  $\theta = 0$  and  $\theta = \pi/2$  are the limitation solutions for these equations.  $\theta = 0$  and  $\theta = \pi/2$  imply that the system will experience three types of LC equilibrium states. When the magnetic field does not reach the MFT threshold point of the LC cell, the uniform solution ( $\theta = 0$ ) shows that the LC molecules remain in the original alignment, in which the total free energy is zero. As the magnetic field increases to the threshold point, the LC cell undergoes a second-order structural phase transition, known as the Magnetic Fréedericksz Transition (MFT), and

$$H_{\text{th}} = (\pi/d)(k_{33}/\chi_a)^{1/2} \quad (5)$$

is the MFT threshold. As the magnetic field goes over the threshold point, the distortion solution ( $\theta = \theta(z)$ ) of the free energy of the system can be written as follows:

$$f_{\theta} = \frac{1}{2} \chi_a H^2 \sin^2 \theta_m - 2\chi_a H^2 \int_0^d \sin^2 \theta dz + W_2 \sin^2 \theta_0 \quad (6)$$

herein, we define  $\theta = \theta_m$  at  $z = \frac{d}{2}$ , and  $\theta_m \approx \left[ \left( \frac{H}{H_0} \right)^2 - 1 \right] / 2B$ ,  $B = (1 - k - 9u/4)/4$ ,  $u = 1 - \frac{\chi_{\perp}}{\chi_{\parallel}}$ ,  $H_0 = \left( \frac{\pi}{d} \right) \left( \frac{K_{33}}{\chi_a} \right)^{1/2}$ .

As the magnetic field continues to increase to a saturation point ( $\theta = \pi/2$ ), the LC molecules will re-orient to the magnetic field director with the following free energy:

$$f_s = -\frac{1}{2} \chi_a H^2 + W_2 \sin^2 \theta_{\text{sat}} \quad (7)$$

In our experiment, we focused on the moment when the magnetic field was around the threshold point of the system. Then,  $f_{\theta}$  is only determined by  $H$ . Thus, we can plot a function curve for the change in the distortion free energy  $f_{\theta}$  of the LC cell with  $H$  based on first two parts of eqn (6). The third part of the anchoring energy can be obtained from the MFT experiment of the pure LCs.

## C. Free energy for anisotropic GNR dispersions in NLC

For free energy theory of anisotropic GNR dispersions in NLC, the free energy per unit volume  $f = F/V$  of the binary compositions has three contributions:

$$f = f_{\text{NLC}} + f_{\text{GNR}} + f_{\text{GNR/NLC}} \quad (8)$$



$f_{\text{NLC}}$  is the free energy density of the NLC,  $f_{\text{GNR}}$  originates from the GNR dispersed in a fluid, and  $f_{\text{NLC/GNR}}$  is obtained from the coupling effect of the two components.

Considering the coupling interaction between the GNR and the NLC host and inspired by Paul van der Schoot and coworker's method for carbon nanotubes dispersed in LC, the coupling free energy per unit volume is in the form of a Boltzmann description:<sup>23</sup>

$$f_{\text{GNR/NLC}} = -\frac{4W}{3D}\phi S_{\text{GNR}} = -\gamma\phi S_{\text{NLC}}S_{\text{GNR}} \quad (9)$$

$\phi$  is the volume fraction,  $\gamma$  is the coupling parameter, and the anchoring energy of LCs for different surfactants is  $W \approx 10^{-5}$  to  $10^{-7}$  N m<sup>-2</sup>.  $D$  is the average cross section diameter of the GNRs.  $S_{\text{NLC}}$  and  $S_{\text{GNR}}$  are the NLC and GNR order parameters. Different types of NLC and GNR with alternative surfactants decide the order parameter. In our experimental setups hereinafter,  $S_{\text{NLC}} \approx 0.6$ , which is a typical order parameter for 5CB at 25 °C,<sup>24</sup> and  $S_{\text{GNR}} \approx -0.2$ , which is estimated using the peak absorbance values of the longitudinal surface plasmon resonance (SPR) of the GNR when the probe light polarization is parallel or perpendicular to the LC director.

According to eqn (9), the coupling parameter has the following relation with  $S_{\text{NLC}}$ ,  $W$ , and  $D$ :<sup>23</sup>

$$\gamma = 4S_{\text{NLC}}^{-1}W/3D \quad (10)$$

Then, by substituting all three terms for free energy into the GNR-LC condition and with quantitative value calculation, we can derive that

$$\frac{df_{\text{GNR/NLC}}}{d\phi} \gg \frac{df_{\text{NLC}}}{d\phi}, \frac{df_{\text{GNR}}}{d\phi} \quad (11)$$

This means that as the doping concentration increases, the change in the coupling free energy takes the main role in the free energy change of the GNR-NLC mixture.

For doped-GNRs in the limited LC cells, the macroscopic distortion solution of the free energy will add an additional term that involves the coupling effect between the GNRs and LC molecules. We have already defined eqn (9) as the expression for the coupling term for GNR/LC per unit. Then, herein, for per area at the substrate, this term can be written as  $f_{\text{GNR/NLC}} = s\gamma\phi d$ , in which  $s = -S_{\text{NLC}}S_{\text{GNR}}$ ,  $\gamma$  is the coupling coefficient,  $\phi$  is the volume fraction, and  $d$  is the thickness of the LC cell. Therefore, the distortion free energy of the doped-GNR LC cell can be written as follows:

$$\begin{aligned} f_{\text{total}} &= f_{\text{elastic+magnetic}}(H, d) + W_{\text{anchoring}} + f_{\text{CNT/NLC}} \\ &= \frac{1}{2}\chi_a H^2 \sin^2 \theta_m - 2\chi_a H^2 \int_0^d \sin^2 \theta dz + W_2 \sin^2 \theta_0 + s\gamma\phi d \end{aligned} \quad (12)$$

Eqn (12) indicates that the extra term for the coupling interaction is similar to an additional anchoring effect for the doped GNR LC cell, and more doped GNR will require a higher magnetic field to reach the MFT threshold of the LC cell. It is also worth noting that we only take into account the elastic free energy of the

LC, and we speculate that the LC temperature is constant; thus, the change in the LC thermotropic free energy density can be ignored.

### III. Experimental

#### A. Synthesis of the gold NRs

Seeded growth of colloidal GNR utilizing binary surfactant mixtures of cetyltrimethylammonium bromide (CTAB) and sodium oleate (NaOL) can improve the dimensional tunability and size uniformity of the GNRs.<sup>25</sup> At first, for the preparation of the seed solution, 5 mL of 0.5 mM HAuCl<sub>4</sub> was mixed with 5 mL of 0.2 M CTAB solution in a 20 mL scintillation vial. Fresh 0.01 M NaBH<sub>4</sub> was diluted in 1 mL ice water and then injected into the Au(III)-CTAB solution. The solution changed from yellow to brownish yellow. For the growth solution, 7.0 g of CTAB and 1.2 g of NaOL were dissolved in 250 mL of warm water (50 °C). After the solution cooled down to 30 °C, 4 mM AgNO<sub>3</sub> solution was added. The mixture was kept undisturbed at 30 °C for 15 min after 250 mL of 1 mM HAuCl<sub>4</sub> solution was added. The solution became colorless after 90 min of stirring (700 rpm), and 1.5 mL of HCl (37 wt% in water) was then introduced to adjust the pH. After another 15 min of slow stirring at 400 rpm, 1.25 mL of 0.064 M ascorbic acid (AA) was added, and the solution was vigorously stirred for 30 s. Finally, 0.8 mL of seed solution was injected into the growth solution. The resultant mixture was stirred for 30 s and left undisturbed at 30 °C for 12 h for NR growth. The final products were isolated by centrifugation at 7000 rpm twice for 10 min to remove the supernatant.

#### B. Synthesis of DMOAP-capped GNRs and doping of GNRs into 5CB LC

To realize the surface coating of DMOAP (*N,N*-dimethyl-*N*-octadecyl-3-aminopropyltrimethoxysilyl chloride), the GNRs needed a two-step method for their prior coating with silica.<sup>26</sup> First, 0.1 M NaOH was added (under vigorous stirring) to 1 mL aqueous dispersion of GNRs with a longitudinal mode optical density (O.D.) of 10. Then, 50  $\mu$ L of 2.5 vol% tetraethyl orthosilicate (TEOS) in ethanol was added under rapid stirring for 0.5 h, and the mixture was kept undisturbed for 24 h. After washing and centrifuging twice at 6000 rpm for 10 minutes, the GNR solution was diluted to 4 mL, and then, 400  $\mu$ L of DMOAP was injected under stirring. The solution was kept stirring for 20 min, washed by centrifugation at 5000 rpm for 8 minutes twice to eliminate the excess DMOAP, and transferred into methanol. Capped-DMOAP GNRs were redispersed in methanol and mixed with 200  $\mu$ L of nematic 4-cyano-4'-pentylbiphenyl (5CB), followed by full evaporation of the solvent within 1 h at 90 °C. By mixing and sonicating for 5 min at 40 °C to put LC in an isotropic state and then vigorously stirring the mixture while cooling down, an excellent dispersion of GNR in the LC phase was achieved.

#### C. LC cell fabrication

K9 glass slides modified with DMOAP as the alignment surfactant were used as substrates for perpendicularly aligned



LC cell. K9 glass slides with the dimensions of  $32 \times 25 \times 0.5$  mm were sonicated for 5 min by acetone, ethanol, and deionized water. The clean K9 glass slides were dipped in the DMOAP (wt 1% in water) aqueous solution for 30 min to assure a homeotropic LC alignment. After washing with deionized water to remove the redundant surfactant, the deposited-DMOAP slides were separated by a 100  $\mu\text{m}$  thick mylar spacer and sealed with epoxy glue. Then, the GNR-LC composite was injected into the cell by capillary forces. We fabricated the LC cells with the doped weight ratios of 0.5%, 1%, 1.5%, 2%, and 5%. Formation of the nematic liquid crystal phase was confirmed by observing birefringence between crossed polarizers. The room temperature was maintained at  $25(\pm 2)$   $^{\circ}\text{C}$  to keep the 5CB in the nematic phase.

#### D. Characterization

Steady-state ultraviolet-visible (UV-vis) absorption spectra of GNRs were obtained using a spectrophotometer (Shimadzu UV-3100). Both the spectra of DMOAP-capped GNR in water and those of the doped-GNR LC cells with different doping weight ratios were obtained. From the absorption spectrum, we can ensure that the orientation of the nanorods was orthogonal to the nematic LC director; thus, a strong longitudinal surface plasmon resonance (LSPR) at 810 nm was triggered by the probe light. Experiment shows that the LSPR redshifts when the GNRs are in the LC; this indicates that the GNRs experience a higher surrounding refractive index in LC than that in water. When the weight ratio reached 5%, the spectra showed a broad saturation absorbance around 810 nm, and the FWHM of the LSPR peak broadened; this meant that a high number of the GNRs in LC had broken the ordered alignment, and scattering was strongly enhanced. Transmission electron microscopy (TEM) was performed using a JEOL 2000FX microscope to determine the morphology of the GNRs, as shown in Fig. 2(b). GNRs with an average length of 69.9 nm and diameter of 23.4 nm have an LSPR wavelength around 810 nm.

#### E. Magnetic Kerr Effect (MKE) experimental setup

The MFT of an LC cell is probed by optical birefringence, as shown in Fig. 3(a). A linearly polarized 632.8 nm He-Ne laser was used as the probe light, with an intensity of 0.05 mW and spot diameter of 26  $\mu\text{m}$  that caused almost no effect on the LC molecules. The LC cell is set between orthogonal polarizers. When the MFT occurs in the LC cell, the dielectric anisotropy of the LC will lead to LC cell birefringence, and residual light passes through. The transmitted probe beam is collected by a photon detector (silicon photodiode) connected to an oscilloscope (TechtronixTDS3012).

## IV. Result and discussion

The experimental measurement of the MFT threshold from the oscilloscope is shown by dots in Fig. 4(a). When the magnetic field increases to the MFT threshold of the LC cell, the LC

molecules will try to rotate in response to the external stimulus, and the LC birefringence will cause a significant increase in the transmitted light; hence, the MFT threshold can be obtained. We notice that as the doping ratio increases, the MFT threshold value increases dramatically. When the doping weight ratio reaches 2%, the MFT threshold is almost 3.5 times that of the pure LC cell; this significant increase is mainly due to the coupling interaction between the GNRs and the LC molecules.

As abovementioned, before  $H$  reaches the MFT threshold, the system is under uniform conditions, and its uniform free energy  $f_{\text{total}} = 0$ . In addition, based on eqn (12), we know the first two part of the distortion free energy exactly, and the third part of anchoring and fourth part of coupling do not depend on  $H$ . Hence, in Fig. 4(a), we can confirm the curves of distortion free energy  $f_{\text{total}}$  with the general shapes predicted by eqn (12) by crossing these curves through the MFT threshold points (significant signal rise positions); this indicates that the LC cell starts to undergo the magnetic Fréedericksz transition at this point. Moreover, from eqn (12), the distortion free energy  $f_{\text{total}}$  at  $H = 0$  refers to only the anchoring term and coupling term. Therefore, we used Fig. 4(b) to plot  $f_{\text{total}}(H = 0)$  versus doping concentration. Since the anchoring term is independent of the doping ratios, we can fit the anchoring and coupling terms with the doping ratios as

$$f_{H=0} = W_{\text{substrate anchoring}} + s\gamma\phi d \quad (13)$$

With the increasing doping ratios, the coupling of the GNR to the LC host fluid also increases, and the free energy of the system undergoes a linear growth before reaching a saturated doping ratio (in our experiment 5% weight ratio is over-saturation). By linearly fitting the points, the slope of the line comes out as  $s\gamma d$ . The thickness of the LC cell is maintained at 100  $\mu\text{m}$  such that the coupling coefficient  $\gamma = 1.50 \times 10^3 \text{ J m}^{-3}$ . According to eqn (10) ( $\gamma = 4S_{\text{NLC}}^{-1}W/3D$ ), the anchoring energy at the GNR surface  $W$  depends on the coupling coefficient  $\gamma$  and the diameter of GNR. Thus, we can calculate the GNR surface anchoring energy to be  $W = 1.35 \times 10^{-5} \text{ J m}^{-2}$ , indicating that DMOAP provides strong anchoring conditions for the GNR-LC interface. On the other hand, the intercept of the fitting line gives  $W_{\text{substrate anchoring}} = 1.05 \times 10^{-5} \text{ J m}^{-2}$ , which agrees with our independent measurement of  $W_{\text{substrate anchoring}} = 1.02 \times 10^{-5} \text{ J m}^{-2}$  based on

$$\frac{H_{\text{th}}}{H_0} \tan \left[ \frac{\pi}{2} \left( \frac{H_{\text{th}}}{H_0} \right) \right] = \frac{d W}{\pi K_{33}} \quad (14)$$

in which  $H_0 = (\pi/d)(K_{33}/\chi_a)^{1/2} = 836 \text{ Oe}$ . This implies that GNRs with a silica coating and DMOAP capping layer have a similar surface anchoring strength as LC anchored on the DMOAP-K9 glass.

The LC response time measurements can provide an independent measure value of the MFT threshold. The response time of an LC cell depends on the external applied field and can be obtained from the dynamic time response of transmitted light from crossed polarizers.<sup>27</sup> Under the conditions of our experiments, the nematic LCs are vertically aligned (VA) on the surface, and zero pretilt angles are set up at the surface





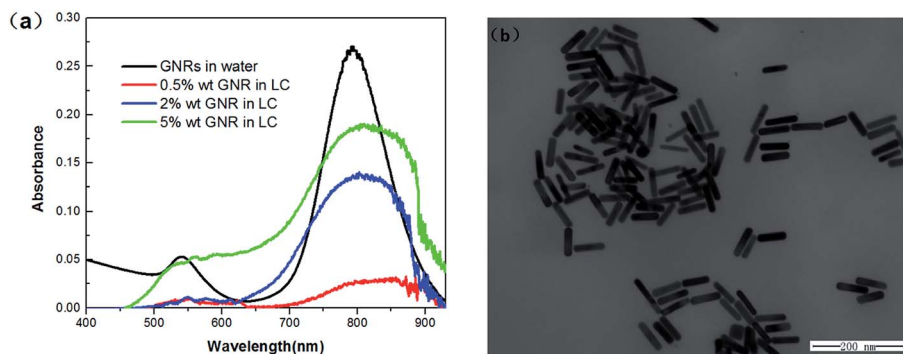


Fig. 2 (a) Absorption spectra of GNR in water or doped at various concentrations in a 5CB LC cell. (b) The TEM image of GNR deposited on a copper net.

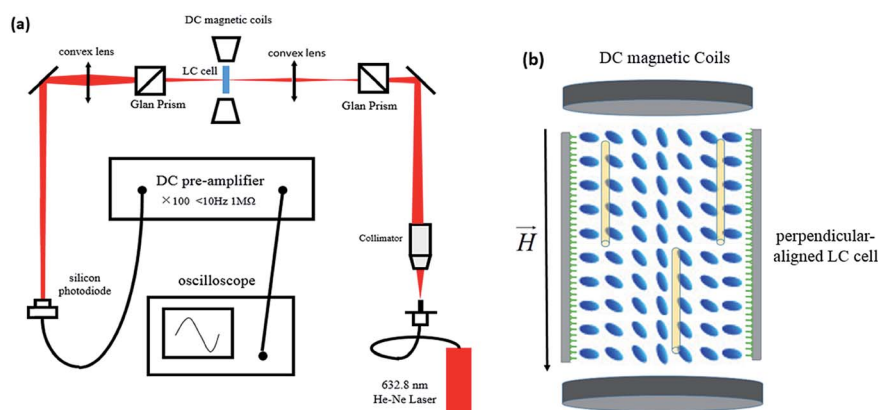


Fig. 3 (a) Setup for the Magnetic Kerr Effect (MKE) experiment. (b) Schematic of an LC cell under a magnetic field.

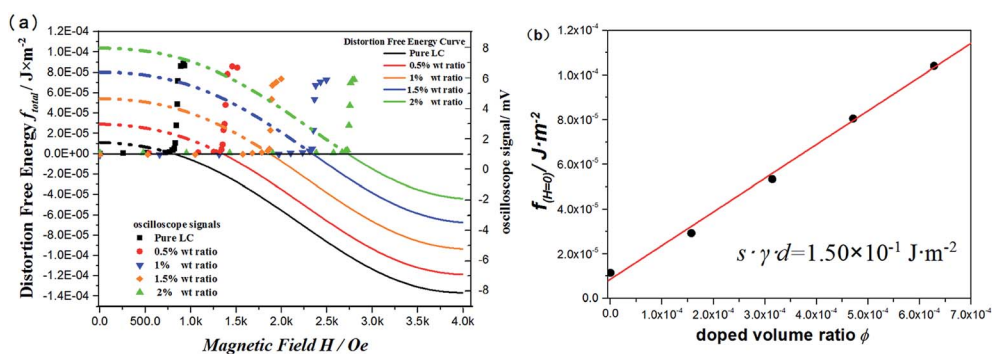


Fig. 4 (a) Dependence of both the induced optical birefringence signals and the simulated free energy  $f_{\text{total}}$  with the pump intensity  $H$  (dotted lines mean that the LC is not under distortion conditions). (b) Free energy (when  $H = 0$ ) as a function of the doping volume ratio  $\phi$ .

boundaries. Supposing that a VA cell is initially biased with a magnetic field ( $H$ ) that is not too far above  $H_{\text{th}}$  and according to the Erickson–Leslie equation, the transmittance  $I(t)$  is

$$I(t) = \sin^2 \left[ \frac{\frac{\delta_0}{2}}{1 + \left( \frac{\phi_\infty^2}{\phi_0^2} - 1 \right) \exp\left(-\frac{2t}{\tau_r}\right)} \right] \quad (15)$$

Upon removing the magnetic field ( $H$ ), the transmittance during the LC cell decay process is

$$I(t) = \sin^2 \left[ \frac{\delta_0 \exp\left(-\frac{2t}{\tau_0}\right)}{2} \right] \quad (16)$$

In these equations,  $\tau_0$  is the LC director reorientation decay time and has the form  $\tau_0 = \gamma_1 d^2 / \pi^2 K_{33}$ , where  $\gamma_1$  is the rotational viscosity, and  $\tau_r$  is the LC director reorientation rise time:



$$\tau_r = \tau_0 / \left[ \left( \frac{H}{H_{th}} \right)^2 - 1 \right] \quad (17)$$

$\delta_0$  is the net phase change,  $\theta_\infty$  is the steady-state tilt angle of the LC directors in response to the applied  $H$ , and  $\theta_0$  is the initial director fluctuation. Eqn (16) can be re-written as

$$\sqrt{\frac{\tau_0}{\tau_r} + 1} = \frac{1}{H_{th}} H \quad (18)$$

During the measurements, the LC cell is optically pumped when a specific  $H$  is applied. The response times are obtained using a digital multi-meter connected to the photodiode. The rise and decay process is measured by applying and removing the magnetic field.  $\tau_r$  and  $\tau_0$  can be obtained by fitting the measured time  $t$  into eqn (15) and (16) for either the rise or the decay process. In Fig. 5(a), the rise and decay process of a 0.5% doped LC cell is given as an example. In each doped sample,  $\tau_0$  does not change with  $H$ , whereas  $\tau_r$  depends on  $H$ . As an example, Fig. 5(b) plots the relation of  $(\tau_0/\tau_r + 1)^{1/2}$  with  $H$  for a 0.5% doped LC cell; other LC cells will have similar linear dependences with a higher magnetic field  $H$ . Therefore, an estimated MFT threshold is deduced for every LC cell. The outcomes are listed in Table 1. We can see that the estimated threshold  $H_{th}$  agrees well with the previous direct measurement results shown in Fig. 4(a). It is also noteworthy that higher doping ratios of GNR in LC show a longer decay time  $\tau_0$ , which can be attributed to the LC order parameter reduction and higher rotational viscosity  $\gamma_1$ .

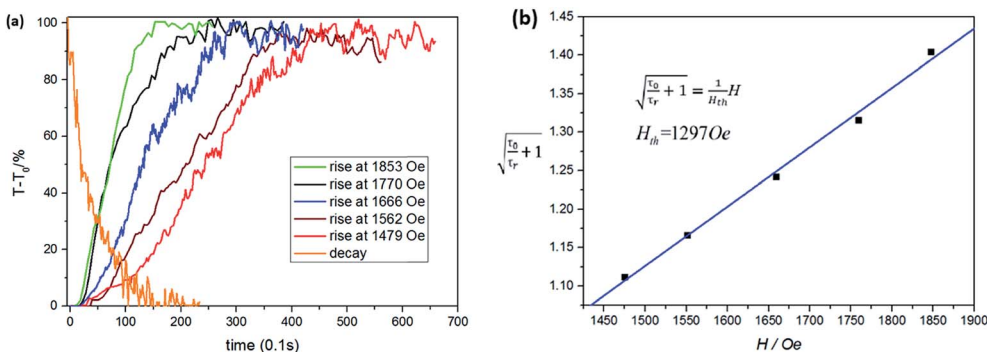
Similar to the DMOAP-capped GNRs, gold nanomaterials of other shapes or surfactants can self-assemble in 5CB LC at high concentrations (herein, GNS is at most 0.5% wt and PEG-GNR at 2% wt). We prepared LC dispersions of gold nanospheres (GNS) and PEG-capped gold nanorods (PEG-GNR) to further reveal the nature of the coupling effect. The preparation of GNS in 5CB LC is carried out according to S. Khatua's methodology,<sup>28</sup> in which 6 nm gold nanospheres form highly stable solutions in 5CB LC by functionalization with 4-sulfanylphenyl-4-[4(octyloxy)phenyl] benzoate (SOPB). Moreover, ordered PEG-GNR dispersions in LC with tangential boundaries have been realized by following

**Table 1** Overall response time measurement result compared with previous  $H_{th}$

LC cell/weight ratio	$\tau_0/s$	Estimated threshold $H_{th}/Oe$	Previous threshold $H_{th}$ in Fig. 4(a)/Oe
Pure	11.8	852	831
0.5%	12.7	1297	1265
1%	14.6	1892	1825
2%	17.9	3015	2910

the Liu's work.<sup>29</sup> The synthesis of GNRs is identical to that of DMOAP-GNRs to maintain the same shapes and initial conditions. By choosing the PEG-SH terminal groups, the main axis of the gold nanorods will show the same orientation as the LC director. We have provided the UV-vis absorption spectra for all types of doped LC cells in Fig. 6(a). It is found that the GNSs have a strong SPR at 550 nm, which is due to their 6 nm spherical shape. The PEG-GNRs have a strong transverse SPR at 560 nm and a weak longitudinal SPR at 810 nm. This means that in a vertical-aligned LC cell, the main axis of the GNR tends to be perpendicular to the substrate and shows little longitudinal SPR.

We used the same experimental setup for the MFT measurements of the LC cell with both GNSs and PEG-GNRs. The outcome is demonstrated in Fig. 6(b). The presence of various nanoparticles in a liquid crystal has been shown to alter the threshold magnetic field in different ways. An enhancement of the magnetic anisotropy due to 6 nm metallic particles is responsible for the slight decrease trend for the GNS samples. Their small size only causes nanometer-scale defects, and the coupling effect can almost be neglected. The magneto-optical properties of PEG-GNRs show the same increase trend as DMOAP-GNRs. We can apply the same process to obtain the PEG-GNR coupling coefficient  $\gamma = 1.12 \times 10^2 \text{ J m}^{-3}$  and anchoring energy  $W = 9.93 \times 10^{-7} \text{ J m}^{-2}$ . The significant reduction in  $W$  for PEG-GNRs implies that silica and DMOAP two step-capped GNRs provide strong anchoring conditions, whereas PEG-SH, owing to its long carbon chain and tangential boundary, only provides weak anchoring conditions.



**Fig. 5** (a) 0.5% doped LC cell rise and decay process with time-resolved measurement. (b) Typical linear relation between  $(\tau_0/\tau_r + 1)^{1/2}$  and the magnetic field  $H$  for a 0.5% doped LC cell.



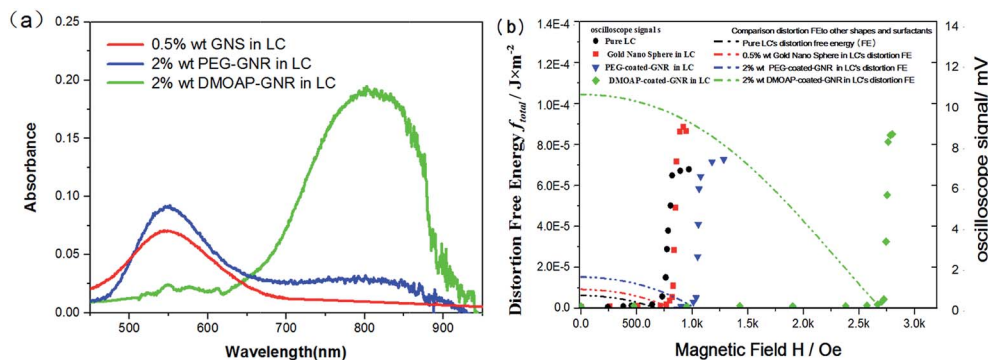


Fig. 6 (a) Absorption spectra of gold nanomaterials doped in a 5CB LC cell. (b) Dependence of induced optical birefringence signals and simulated free energy  $f_{\text{total}}$  with the pump intensity  $H$ .

## V. Conclusion

In summary, we have successfully developed an approach to measure the anisotropic anchoring strength of NRs in an LC host. As a typical case, by utilizing perpendicular alignment of GNR dispersed in a thermotropic LC and measuring the MFT thresholds of the LC cell, the coupling coefficient of the GNR-doped LC is deduced. Combining the Landau-de Gennes free energy theory of the binary mixture of GNR and LC, an internal view of LC and NR coupling interaction is obtained, and the anchoring energy between the LC and NR can be derived. The response time data of the doped LCs has been provided as additional confirmation for the threshold measurement. This method can be further extended to other nanorod-LC systems (such as CdS<sup>30</sup> or CNT) to derive their coupling coefficients. The properties of binary GNR-LC mixtures can provide a great potential for the design and realization of new devices<sup>28,29</sup> for guest-host optoelectronics and display applications.

## Conflicts of interest

There are no conflicts to declare.

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## References

- J. A. Fan, C. Wu, K. Bao, J. Bao, R. Bardhan, N. J. Halas, V. N. Manoharan, P. Nordlander, G. Shvets and F. Capasso, *Science*, 2010, **328**, 1135–1138.
- J. Kao, K. Thorkelsson, P. Bai, B. J. Rancatore and T. Xu, *Chem. Soc. Rev.*, 2013, **42**, 2654–2678.
- J. Shan, W. Shi, L. Y. Liu, Y. R. Shen and L. Xu, *Phys. Rev. Lett.*, 2012, **109**, 147801.
- X. Zheng, J. Fontana, M. Pevnyi, M. Ignatenko, S. Wang, R. Vaia and P. Palffy-Muhoray, *J. Mater. Sci.*, 2012, **47**, 4914–4920.
- X. Huang, S. Neretina and M. A. El-Sayed, *Adv. Mater.*, 2009, **21**, 4880–4910.
- C. J. Murphy, T. K. Sau, A. M. Gole, C. J. Orendorff, J. Gao, L. Gou, S. E. Hunyadi and T. Li, *J. Phys. Chem. B*, 2005, **109**, 13857–13870.
- T. A. Wood, J. S. Lintuvuori, A. B. Schofield, D. Marenduzzo and W. C. K. Poon, *Science*, 2011, **334**, 79–83.
- H. K. Bisoy and S. Kumar, *Chem. Soc. Rev.*, 2011, **40**, 306–319.
- G. M. Koenig, J. J. de Pablo and N. L. Abbott, *Langmuir*, 2009, **25**, 13318–13321.
- V. Tomar, T. F. Roberts, N. L. Abbott, J. P. Hernandez-Ortiz and J. J. de Pablo, *Langmuir*, 2012, **28**, 6124–6131.
- S. Umadevi, X. Feng and T. Hegmann, *Adv. Funct. Mater.*, 2013, **22**, 1393–1403.
- S. Sridevi, S. K. Prasad, G. G. Nair, V. D. Britto and B. L. V. Prasad, *Appl. Phys. Lett.*, 2010, **97**, 151913.
- Y. Zhang, Q. Liu, H. Mandoor, Y. Yuan and I. I. Smalyukh, *ACS Nano*, 2015, **9**, 3097–3108.
- G. H. Sheetah, Q. Liu and I. I. Smalyukh, *Opt. Lett.*, 2016, **41**, 4899–4902.
- B. Senyuk, D. Glugla and I. I. Smalyukh, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2013, **88**, 062507.
- C. W. Hsu, B. Zhen, W. Qiu, O. Shapira, B. G. DeLacy, J. D. Joannopoulos and M. Soljacic, *Nat. Commun.*, 2014, **5**, 3152.
- T. Ellenbogen, K. Seo and K. B. Crozier, *Nano Lett.*, 2012, **12**, 1026–1031.
- P. Schoot, V. Popa-Nita and S. Kralj, *J. Phys. Chem. B*, 2008, **112**, 4512–4518.
- P. G. de Gennes and J. Prost, *The Physics of Liquid Crystals*, Clarendon: Oxford, U.K., 1995, pp. 53–66.
- X. Nie, R. Lu, H. Xian, T. X. Wu and S.-T. Wu, *J. Appl. Phys.*, 2007, **101**, 103110.
- J. P. F. Lagerwall and G. Scalia, *J. Mater. Chem.*, 2008, **18**, 2890.
- H. L. Ong, *Phys. Rev. A*, 1986, **33**, 3550.
- V. Popa-Nita and S. Kralj, *Opt. Lett.*, 2010, **132**, 024902.
- R. G. Horn, *J. Phys.*, 1978, **39**, 105–109.



- 25 X. Ye, C. Zheng, J. Chen, Y. Gao and C. B. Murray, *Nano Lett.*, 2013, **13**, 765–771.
- 26 W. C. Wu and J. B. Tracy, *Chem. Mater.*, 2015, **27**, 2888.
- 27 H. Wang and T. X. Wu, *J. Appl. Phys.*, 2004, **95**, 5502.
- 28 S. Khatua, P. Manna, W.-S. Chang, A. Tcherniak, E. Friedlander, E. R. Zubarev and S. Link, *J. Phys. Chem. C*, 2010, **114**, 7251–7257.
- 29 Q. Liu, Y. Yuan and I. I. Smalyukh, *Nano Lett.*, 2014, **14**, 4071–4077.
- 30 A. A. Ezhov, G. A. Shandryuk, G. N. Bondarenko, A. S. Merekalov, S. S. Abramchuk, A. M. Shatalova, P. Manna, E. R. Zubarev and R. V. Talroze, *Langmuir*, 2011, **27**, 13353–13360.

