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Improving the intrinsic thermal stability of the MAPbI₃ perovskite by incorporating cesium 5-aminovaleric acetate†

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Cesium 5-aminovaleric acetate (NH₂C₄H₈COOCs) was used to improve the intrinsic thermal stability of the methylammonium lead triiodide (MAPbI₃) perovskite. The corresponding carbon-based perovskite solar cells without encapsulation showed favourable stability at 100 °C for 500 h.

Next-generation solar cells have the prerequisite of low-cost and easy to fabricate are the prerequisites of next-generation printable solar cells. In recent years, printable solar cells based on ABX₃ (where A is typically methylammonium (MA), formamidinium (FA), or Cs; B is Pb or Sn; and X is I, Br or Cl) perovskite-type light absorbers have attracted widespread attention for their high power conversion efficiencies (PCEs) combining with low processing costs. At present, the record PCE for any perovskite solar cell (PSC) is 22.7%,¹ which is on a par with those of mainstream multi-crystalline silicon solar devices. Obviously, the high PCE together with easy fabrication procedures make PSCs a captivating device for future applications.

Typically PSCs need a vacuum system to evaporate a noble metal (Ag or Au) as the back contact. However, the use of vacuum system and expensive metal targets with high purity doesn't suit the purpose of low-cost printing. In this regard, the carbon-based perovskite solar cells,² which substitute the noble metal electrode with a printable carbon electrode, have enormous potential for realizing the application of fully printed, low-cost photovoltaics. To date, many groups from all over the world have put much energy into the study of improving the performance of such PSCs with mesoporous carbon electrodes. Strategies including surface modification,^{3–5} materials engineering,^{6–11} solvent engineering^{12–14} and post-treatments^{15–17}

have been applied in carbon-based PSCs for pursuing higher efficiencies. As a result, the reported PCE of carbon-based PSCs has increased rapidly from 6.6%² to 17%¹⁸ in just a few years. And, at the same time, carbon-based PSCs have been reported that can be easily printed into large-scale modules.¹⁹ However, just like any other kind of PSC, long-term stability is still a problem for the final commercialization of this emerging photovoltaic technology.²⁰ Solar modules are exposed to elevated temperatures during operation; hence as per the international standard (IEC 61646 climatic chamber tests), a solar panel must show thermal stability up to 85 °C.²¹ The formation energy for the MAPbI₃ perovskite is 0.11–0.14 eV, which is close to 0.093 eV, suggesting possible degradation at a continuous exposure to a temperature of 85 °C. Recently, formamidinium cation (FA⁺)- and Cs⁺- based perovskites have demonstrated better thermal stability levels than has pure MAPbI₃. However, FAPbI₃ and CsPbI₃ perovskites are more sensitive to humidity, which requires higher costs for encapsulation.^{22,23}

In carbon-based PSCs, perovskite crystals are loaded in the mesopores, and their grain sizes are restricted by the pore size. In comparison to the perovskite crystals with large grains in traditional PSCs, perovskite crystals with small grains in carbon-based PSCs show much poorer heat tolerance. For this reason, all of the carbon-based PSCs using an MAPbI₃ perovskite light absorber can only be annealed at 50–70 °C.^{2,4,8,16,19} From this perspective, enhancing the intrinsic heat tolerance of perovskite materials would be of great significance for the development of carbon-based PSCs.

Herein, we used cesium 5-aminovaleric acetate (NH₂C₄H₈COOCs) as an additive of MAPbI₃ perovskite, and found that Cs_xMA_{1–x}Pb(5-AVA)_xI_{3–x} perovskite showed high heat tolerance, even at 150 °C. Based on previous work,⁸ 5-ammonium valeric acid (5-AVA) hydriodide (NH₂C₄H₈COOH·HI), which was obtained from acid (HI) hydrolysis of 5-AVA, could create mixed-cation perovskite (5-AVA)_x(MA)_{1–x}PbI₃ crystals with preferable

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stability in ambient air under full sunlight. Since 5-AVA is an amphipathic molecule, we employed CsOH to synthesize the alkaline hydrolysis product of 5-AVA (see ESI† for the synthesis details), aiming at combining the positive effects of the 5-AVA group and Cs⁺ cation in the MAPbI₃ perovskite.

The molecular structures of (5-AVA) iodide and Cs-(5-AVA) acetate are shown in Fig. 1a. *Via* replacing 5% (in molar ratio) of MAI in the MAPbI₃ perovskite with (5-AVA) iodide and Cs-(5-AVA) acetate, we obtained the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite and Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite, respectively. Note that 5-AVA served as a cation in the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite, while in the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite, 5-AVA took the place of the I⁻ anion. This little difference resulted in different solubilities of the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite and Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite in γ -butyrolactone (GBL) solvent. As shown in Fig. 1b, the turbid (5-AVA)_x(MA)_{1-x}PbI₃ perovskite solution (1 M in GBL) showed a lot of yellowish precipitate at room temperature. However, the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite solution was much clearer, indicating the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite to be more soluble than the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite. We attributed the good solubility of Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite in GBL to the exposed amino group of the Cs-(5-AVA) acetate.

X-ray diffraction (XRD) measurements were taken to identify the crystal structure of the as prepared perovskites. On glass substrate, both of the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} and (5-AVA)_x(MA)_{1-x}PbI₃ perovskite patterns showed similar diffraction peaks at 14.18°, 23.5°, 24.5°, 28.56° and 31.05°, indicating the same tetragonal MAPbI₃ perovskite crystal structure.²⁴ What's more, the intensities of the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} peaks were much stronger than those of the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite peaks. Since the two kinds of perovskite samples were fabricated and measured using the same conditions, we concluded the Cs-(5-AVA) acetate to be beneficial for the growth of the MAPbI₃ perovskite (Fig. 2).

By using these two kinds of perovskite materials, we fabricated carbon-based PSCs according to the method reported previously.² Briefly, by using the screen printing technique, mesoporous TiO₂, ZrO₂ and carbon films were successively deposited on the FTO substrates, which had been coated with compact TiO₂ beforehand. And then, a perovskite light absorber was loaded by filling the precursor solution into the mesoporous carbon/ZrO₂/TiO₂ layers (see Fig. 3a). The

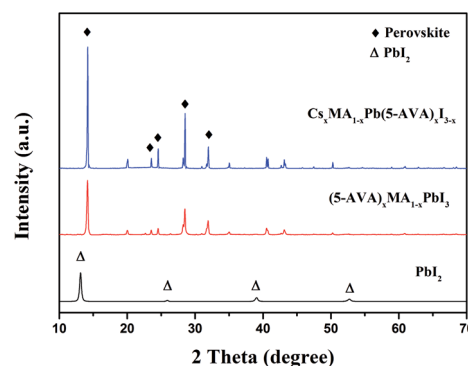


Fig. 2 XRD patterns of the PbI₂, (5-AVA)_x(MA)_{1-x}PbI₃ and Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} films on glass substrates.

microstructure of a cross-section of the as-prepared device was observed using SEM, and each layer showed well-defined boundaries and a uniform thickness (see Fig. 3b), and the distribution of the perovskite in the mesoporous films was homogenous (EDS images are shown in Fig. S1†). After the filling procedure, the devices were dried on a hot plate, and after the removal of GBL solvent, we finally obtained the carbon-based PSCs. For the sake of comparison, the drying temperature was set at 100 °C, which was much higher than the conventionally used temperature⁸ (50 °C).

Two groups of the carbon-based PSCs (with each group consisting of eight devices) were fabricated by using the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} and (5-AVA)_x(MA)_{1-x}PbI₃ perovskites, respectively. Under standard AM1.5 illumination (100 mW cm⁻²), the devices using the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite generally showed a higher PCE than did the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite. (Details of the photovoltaic parameters are summarized in Table S1.†) *J*-*V* curves of the corresponding best devices are shown in Fig. 4a. With the Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite, the best device showed a photocurrent density (*J*_{sc}) of 20.59 mAcm⁻² (with *J*_{sc} determined from the IPCE spectra in Fig. S2†), an open circuit voltage (*V*_{oc}) of 893 mV, a fill factor (FF) of 0.66 and an overall PCE of 12.19%. In contrast, the best device using the (5-AVA)_x(MA)_{1-x}PbI₃ perovskite showed a much lower PCE of 9.50% (*J*_{sc} = 16.78 mAcm⁻², *V*_{oc} = 830 mV and FF = 0.68). Moreover, the long-term stability levels of the best devices without encapsulation were determined at 100 °C in a glove box (see Fig. 4b). After 500 h, the PCE of the (5-AVA)_x(MA)_{1-x}PbI₃-based device decayed gradually from 9.50%

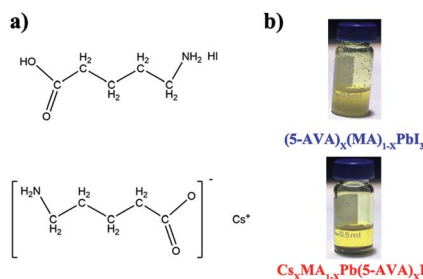


Fig. 1 (a) The molecular structure of (5-AVA) iodide and Cs-(5-AVA) acetate. (b) Optical images of (5-AVA)_x(MA)_{1-x}PbI₃ and Cs_xMA_{1-x}Pb(5-AVA)_xI_{3-x} perovskite solution at room temperature.

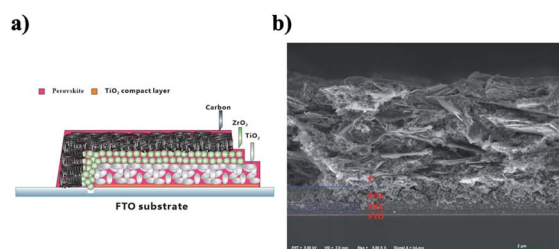


Fig. 3 (a) Schematic structure of the carbon-based PSC. (b) SEM image from a cross-section of the carbon-based PSC.



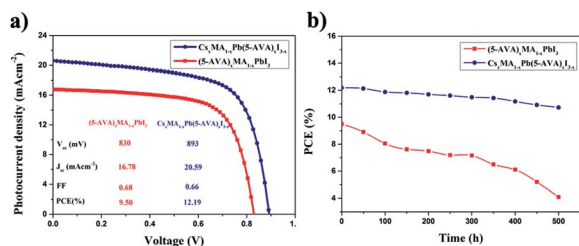


Fig. 4 (a) *J*-*V* curves of the carbon-based PSCs with the (5-VA)_x(MA)_{1-x}PbI₃ and Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskites. (b) Long-term stability of the carbon-based PSCs stored at 100 °C in glove box.

to 4.1%, *i.e.*, a reduction of 56.8%. Surprisingly, the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x}-based device maintained 88% (decay from 12.19% to 10.73%) of its initial PCE after 500 h, indicating favourable thermal stability at 100 °C.

Further XRD measurements were taken to confirm the difference between the thermal stability of the (5-VA)_x(MA)_{1-x}PbI₃ perovskite and that of the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite. The (5-VA)_x(MA)_{1-x}PbI₃ perovskite, after having been heated on a hot plate at 75 °C for 24 h, yielded an XRD pattern showing a small peak at 2θ = 12.63°, indicating that part of the perovskite was decomposed into PbI₂. As the temperature was increased, the intensity of the PbI₂ diffraction peak became increasingly stronger, and the colour of (5-VA)_x(MA)_{1-x}PbI₃ perovskite film on glass gradually changed from black to yellow (see Fig. 5a). However, compared with the (5-VA)_x(MA)_{1-x}PbI₃ perovskite, the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite displayed a much lower thermal decomposition rate. As shown in Fig. 5b, even after having been heated at 150 °C for 24 h, most of the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite still kept the perovskite phase (see Fig. 5b).

To further identify any possible effects of the (5-VA) anion and Cs cation on the thermal stability of the MAPbI₃ perovskite, we also compared the decomposition process of MAPb(5-VA)_xI_{3-x} with that of Cs_xMA_{1-x}PbI_{3-x} (see Fig. S3†). The XRD results showed that including 5% Cs⁺ individually or 5% (5-VA)⁻ individually in the MAPbI₃ perovskite sample did not much enhance the thermal stability of the sample. Hence, we attributed the good thermal stability of the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite to the unique Cs(5-VA) acetate additive. Since the 5-VA group can act as a templating agent in

perovskites^{8,25,26} and the Cs⁺ cation has a strong bonding energy with the I⁻ anion, the migration of ions in the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite could be suppressed by the 2D/3D interfaces²⁷ and the high migration activation energy of the inorganic cation.²⁸

Conclusions

In summary, we have developed a novel Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite by incorporating the Cs(5-VA) acetate into the MAPbI₃ perovskite. Carbon-based PSCs using such Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskites exhibited a favourable PCE of 12.19%. Moreover, after having been stored at 100 °C in a glove box for 500 h, the device still maintained 88% of its initial PCE. The XRD results showed the Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite to have a thermal decomposition rate much lower than that of the mixed-cation (5-VA)_x(MA)_{1-x}PbI₃ perovskite. Combining Cs⁺ with an alkyl chain to suppress the ion migration in MAPbI₃ perovskites could be a new strategy for enhancing the thermal stability of PSCs.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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Notes and references

- 1 N. R. E. L. N. Best Research Cell_Efficiencies, October. 2017, <https://www.nrel.gov/pv/assets/images/efficiency-chart.png>.
- 2 Z. L. Ku, Y. G. Rong, M. Xu, T. F. Liu and H. W. Han, *Sci. Rep.*, 2013, **3**, 3132.
- 3 J. Ryu, K. Lee, J. Yun, H. Yu, J. Lee and J. Jang, *Small*, 2017, **13**, 1701225.
- 4 F. Behrouznejad, C.-M. Tsai, S. Narra, E. W. G. Diao and N. Taghavinia, *ACS Appl. Mater. Interfaces*, 2017, **9**, 25204–25215.
- 5 L. Liu, A. Mei, T. Liu, P. Jiang, Y. Sheng, L. Zhang and H. Han, *J. Am. Chem. Soc.*, 2015, **137**, 1790–1793.
- 6 V. Kapoor, A. Bashir, L. J. Haur, A. Bruno, S. Shukla, A. Priyadarshi, N. Mathews and S. Mhaisalkar, *Energy Technol.*, 2017, **5**, 1880–1886.
- 7 M. Hu, L. Liu, A. Mei, Y. Yang, T. Liu and H. Han, *J. Mater. Chem. A*, 2014, **2**, 17115–17121.

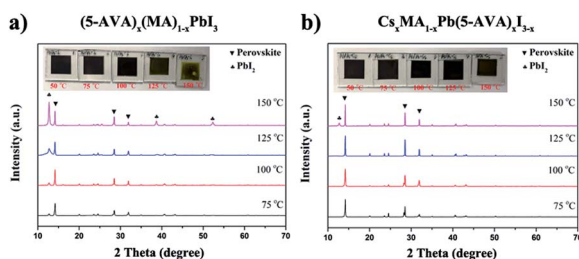


Fig. 5 XRD patterns and optical images (inset) of (a) (5-VA)_x(MA)_{1-x}PbI₃ and (b) Cs_xMA_{1-x}Pb(5-VA)_xI_{3-x} perovskite films exposed on a hot plate to various temperatures each for 24 h.



- 8 A. Mei, X. Li, L. Liu, Z. Ku, T. Liu, Y. Rong, M. Xu, M. Hu, J. Chen, Y. Yang, M. Gratzel and H. Han, *Science*, 2014, **345**, 295–298.
- 9 X. Zheng, H. Chen, Q. Li, Y. Yang, Z. Wei, Y. Bai, Y. Qiu, D. Zhou, K. S. Wong and S. Yang, *Nano Lett.*, 2017, **17**, 2496–2505.
- 10 T. A. N. Peiris, A. K. Baranwal, H. Kanda, S. Fukumoto, S. Kanaya, L. Cojocar, T. Bessho, T. Miyasaka, H. Segawa and S. Ito, *Nanoscale*, 2017, **9**, 5475–5482.
- 11 Y. Rong, X. Hou, Y. Hu, A. Mei, L. Liu, P. Wang and H. Han, *Nat. Commun.*, 2017, **8**, 14555.
- 12 J. Chen, Y. Xiong, Y. Rong, A. Mei, Y. Sheng, P. Jiang, Y. Hu, X. Li and H. Han, *Nano Energy*, 2016, **27**, 130–137.
- 13 C.-Y. Chan, Y. Wang, G.-W. Wu and E. Wei-Guang Diao, *J. Mater. Chem. A*, 2016, **4**, 3872–3878.
- 14 H. Chen, Z. Wei, H. He, X. Zheng, K. S. Wong and S. Yang, *Adv. Energy Mater.*, 2016, **6**, 1502087.
- 15 Y. Zhang, J. Wang, X. Liu, W. Li, F. Huang, Y. Peng, J. Zhong, Y. Cheng and Z. Ku, *RSC Adv.*, 2017, **7**, 48958–48961.
- 16 S. G. Hashmi, D. Martineau, M. I. Dar, T. T. T. Myllymaki, T. Sarikka, V. Ulla, S. M. Zakeeruddin and M. Gratzel, *J. Mater. Chem. A*, 2017, **5**, 12060–12067.
- 17 L. Hong, Y. Hu, A. Mei, Y. Sheng, P. Jiang, C. Tian, Y. Rong and H. Han, *Adv. Funct. Mater.*, 2017, **27**, 1703060.
- 18 S. Liu, W. Huang, P. Liao, N. Pootrakulchote, H. Li, J. Lu, J. Li, F. Huang, X. Shai, X. Zhao, Y. Shen, Y.-B. Cheng and M. Wang, *J. Mater. Chem. A*, 2017, **5**, 22952–22958.
- 19 A. Priyadarshi, L. J. Haur, P. Murray, D. Fu, S. Kulkarni, G. Xing, T. C. Sum, N. Mathews and S. G. Mhaisalkar, *Energy Environ. Sci.*, 2016, **9**, 3687–3692.
- 20 J. S. Shaikh, N. S. Shaikh, A. D. Sheikh, S. S. Mali, A. J. Kale, P. Kanjanaboos, C. K. Hong, J. H. Kim and P. S. Patil, *Mater. Des.*, 2017, **136**, 54–80.
- 21 B. Conings, J. Drijkoningen, N. Gauquelin, A. Babayigit, J. D'Haen, L. D'Olieslaeger, A. Ethirajan, J. Verbeeck, J. Manca, E. Mosconi, F. D. Angelis and H.-G. Boyen, *Adv. Energy Mater.*, 2015, **5**, 1500477.
- 22 B. Li, Y. Li, C. Zheng, D. Gao and W. Huang, *RSC Adv.*, 2016, **6**, 38079–38091.
- 23 T.-B. Song, Q. Chen, H. Zhou, C. Jiang, H.-H. Wang, Y. Yang, Y. Liu, J. You and Y. Yang, *J. Mater. Chem. A*, 2015, **3**, 9032–9050.
- 24 T. Baikie, Y. Fang, J. M. Kadro, M. Schreyer, F. Wei, S. G. Mhaisalkar, M. Graetzel and T. J. White, *J. Mater. Chem. A*, 2013, **1**, 5628.
- 25 A. K. Baranwal, S. Kanaya, T. A. Peiris, G. Mizuta, T. Nishina, H. Kanda, T. Miyasaka, H. Segawa and S. Ito, *ChemSusChem*, 2016, **9**, 2604–2608.
- 26 G. Grancini, C. Roldán-Carmona, I. Zimmermann, E. Mosconi, X. Lee, D. Martineau, S. Narbey, F. Oswald, F. De Angelis, M. Graetzel and M. K. Nazeeruddin, *Nat. Commun.*, 2017, **8**, 15684.
- 27 X. Xiao, J. Dai, Y. Fang, J. Zhao, X. Zheng, S. Tang, P. N. Rudd, X. C. Zeng and J. Huang, *ACS Energy Lett.*, 2018, **3**, 684–688.
- 28 D. Pan, Y. Fu, J. Chen, K. J. Czech, J. C. Wright and S. Jin, *Nano Lett.*, 2018, **18**, 1807–1813.

