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# Influence of citrate on phase transformation and photoluminescence properties in LaPO<sub>4</sub> and LaPO<sub>4</sub>:Eu<sup>3+</sup>

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The hexagonal and monoclinic phase LaPO<sub>4</sub> and LaPO<sub>4</sub>:Eu nanostructures have been controllably synthesized by a citrate-induced hydrothermal process at 100 °C. The crystal growth of LaPO<sub>4</sub> nanostructures was investigated, and the phase transformation of nanostructured LaPO<sub>4</sub> was systematically studied by varying the citrate concentration, pH value and reaction temperature. When 0.8 mmol of citrate was added into the reaction system, the hexagonal phase LaPO<sub>4</sub> transformed into the monoclinic phase. High concentrations of citrate would lead to the formation of hexagonal phase LaPO<sub>4</sub>. The photoluminescence properties of the monoclinic phase LaPO<sub>4</sub>:Eu prepared using a citrate-induced process demonstrate that the electric dipole transition (<sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>2</sub>) is stronger than the magnetic dipole transition (<sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>1</sub>), which indicated that Eu<sup>3+</sup> is in a site with no inversion center. The strongest emission peak of hexagonal phase LaPO<sub>4</sub>:Eu comes from <sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>1</sub>. Furthermore, the citrate-induced hexagonal phase LaPO<sub>4</sub>:Eu has a stronger emission intensity than the hexagonal phase LaPO<sub>4</sub>:Eu prepared not using a citrate-induced process.

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## 1 Introduction

In recent years, the rare earth orthophosphates have received much attention, due to their electronic, optical, and chemical characteristics.<sup>1</sup> For example, they are used as phosphors, light sources, high-performance luminescent devices, field-effect transistors, solar cells, and biomedical labels.<sup>2–7</sup> Rare earth orthophosphates are rich in polymorphs, which usually include monazite (monoclinic), xenotime (tetragonal), rhabdophane (hexagonal), weinschenkite (monoclinic).<sup>8</sup> Phase transformation is very common in these polymorphs. The hexagonal phase rare earth orthophosphates are obtained at low temperature, and can transform into the monoclinic phase at high temperature.<sup>9</sup> As important photoluminescent host materials, the hexagonal structured lanthanum orthophosphates are usually prepared from a sol–gel method<sup>10</sup> and hydrothermal method<sup>11</sup> at low-temperature, while the monoclinic phase lanthanum orthophosphates can be prepared at high temperature. For example, the monoclinic phase LaPO<sub>4</sub> could be synthesized through hydrothermal reaction at 180 °C;<sup>12</sup> the monoclinic phase LaPO<sub>4</sub>:Er was obtained *via* solid state reaction at 1200 °C;<sup>13</sup> and the monoclinic

phase LaPO<sub>4</sub>:Eu could also be formed in high boiling coordinating solvents at 200 °C.<sup>14</sup> Recently, the low temperature synthesis method of the monoclinic phase REPO<sub>4</sub> has received much attention. Several pressure-induced phase transformations of ABO<sub>4</sub> (A = Ba, Ca, Sr, Tb, Dy; B = Cr, W, P) compounds have been discovered.<sup>15–18</sup> In particular, the organic ligand-induced phase transformation of REPO<sub>4</sub> compounds at low temperature has been studied. Yan *et al.*<sup>19</sup> reported an ethylenediaminetetraacetic acid (EDTA)-mediated hydrothermal route to synthesize different phase cerium orthovanadate (CeVO<sub>4</sub>) microcrystals. In our previous study, we found that nanostructured CePO<sub>4</sub>:Tb with hexagonal and monoclinic phases could be controllably synthesized through a hydrothermal route at 150 °C by simply varying the reactant C<sub>2</sub>O<sub>4</sub><sup>2-</sup>/Ce molar ratio.<sup>20</sup> Nuria *et al.*<sup>21</sup> have synthesized monazite LnPO<sub>4</sub> (Ln = La, Ce) and LaPO<sub>4</sub>:Ln (Ln = Eu, Ce, Ce + Tb) through the controlled release of La<sup>3+</sup> cations from lanthanide–citrate complexes using ethylene glycol (EG) as solvent. Among a variety of organic additives, trisodium citrate (Cit<sup>3-</sup>) is one of the most common and important organic molecules.<sup>22,23</sup> Actually, citrate is often used as a structure-directing agent to control the nucleation, growth and alignment of REPO<sub>4</sub> crystals.<sup>24,25</sup> In order to achieve the phase transformation of REPO<sub>4</sub> compounds at low temperature, the organic ligand-induce might be a feasible method.

In this paper, we developed an effective method to synthesis the LaPO<sub>4</sub> and LaPO<sub>4</sub>:Eu with hexagonal phase and monoclinic phase by the citrate-induced hydrothermal process at low temperature. Furthermore, the function of citrate-induce on the phase transformation of LaPO<sub>4</sub> was systematically investigated.

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It is interesting to note that monoclinic phase  $\text{LaPO}_4$  nanoparticles grow better if 0.8 mmol of citrate is used. Similarly, the monoclinic phase and hexagonal phase  $\text{CePO}_4$  can be obtained, respectively. The monoclinic phase  $\text{CePO}_4$  can be obtained while citrate concentration increased to 1.2 mmol. High concentration of citrate would result in the formation of hexagonal phase rare earth orthophosphates. We also discussed the effect of high concentration citrate on the phase transformation, and the photoluminescence properties of the  $\text{LaPO}_4:\text{Eu}$  with different phase and size.

## 2 Experimental

### 2.1 Material and reagents

All chemicals were of analytical grade and were used as received without further purification.  $\text{H}_3\text{PO}_4$  (A.R.),  $\text{HNO}_3$  (A.R.),  $\text{NH}_3 \cdot \text{H}_2\text{O}$  (A.R.),  $\text{La}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$  (A.R.), sodium citrate dihydrate,  $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$  (A.R.) and  $\text{Eu}_2\text{O}_3$  (purity > 99.99%) were all supplied by Sinopharm Chemical Reagent Limited Corporation. The europium nitrate was prepared by dissolving  $\text{Eu}_2\text{O}_3$  in 10% nitric acid, and then evaporated and dried in vacuum.

### 2.2 Synthesis

In a typical procedure using citrate as ligand, 0.0–3.0 mmol sodium citrate dihydrate was added to  $0.1 \text{ mol L}^{-1}$  of  $\text{La}(\text{NO}_3)_3$  solution while kept under stirring. The solution turned turbid due to the formation of lanthanum citrate complex under vigorous stirring for 20 min. Subsequently,  $0.1 \text{ mol L}^{-1}$   $\text{H}_3\text{PO}_4$  was added slowly while kept under stirring at 10 min. The obtained turbid solution was transferred into a stainless steel autoclave with an inner Teflon vessel (volume, 50 mL). It was sealed and maintained at  $100^\circ\text{C}$  for 12 h. After cooling to room temperature, the precipitation was separated by centrifugation, washed with deionized water and ethanol absolute three times, and finally dried at  $60^\circ\text{C}$  for 8 h. In this way, the  $\text{La}_{0.95}\text{PO}_4 \cdot \text{Eu}_{0.05}$  and  $\text{CePO}_4$  were synthesized.

### 2.3 Characterization

The size and morphology of the products were characterized by scanning electronic microscopy (SEM, Hitachi S-4800, Japan) and transmission electron microscopy (TEM, FEI Tecnai F20, USA). XRD patterns were measured by a 21 kW extra power X-ray diffractometer (Model M21XVHF22, MAC science Co. Ltd., Japan) using  $\text{Cu K}\alpha$  radiation ( $k = 0.1541 \text{ nm}$ ) over a  $2\theta$  range of  $10\text{--}60^\circ$  at room temperature. The photoluminescence spectra of powders were recorded on FL Spectrophotometer (FLS-980) with the slit width of 1.0 nm at room temperature.

## 3 Result and discussion

### 3.1 Structure of $\text{LaPO}_4$ and $\text{CePO}_4$

In this study, the different crystalline phases of  $\text{LaPO}_4$  were prepared using different concentration of citrate as ligand. The phase structures of the products were identified by X-ray diffraction (Fig. 1). The typical XRD pattern of the product prepared with 0.8 mmol citrate is shown in Fig. 1a, all

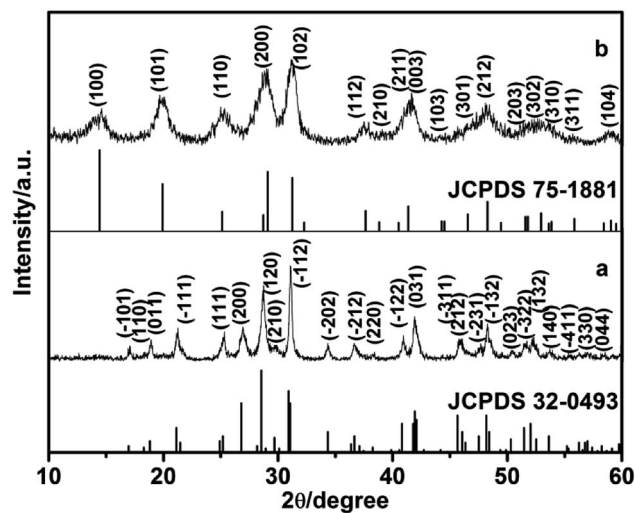


Fig. 1 XRD patterns of  $\text{LaPO}_4$  nanostructures prepared with citrate-induced (a) the monoclinic phase and (b) the hexagonal phase.

diffraction peaks agree well with monoclinic phase  $\text{LaPO}_4$  (JCPDS 32-0493). The diffraction peaks were very strong and sharp, indicating that the sample has a good crystallinity. While citrate concentration is 3.0 mmol, all the peaks can be indexed to the hexagonal phase  $\text{LaPO}_4$ , which was in good agreement with the JCPDS 75-1881 (Fig. 1b). Therefore, the phase transformation of  $\text{LaPO}_4$  can be controlled by varying citrate concentration.

TEM images of as-synthesized hexagonal and monoclinic phase products were shown in Fig. 2. Fig. 2a presents the TEM images of the product synthesized with 0.8 mmol citrate. The as-synthesized monoclinic phase  $\text{LaPO}_4$  was composed of nanoparticles, its average grain size was about 40 nm by using

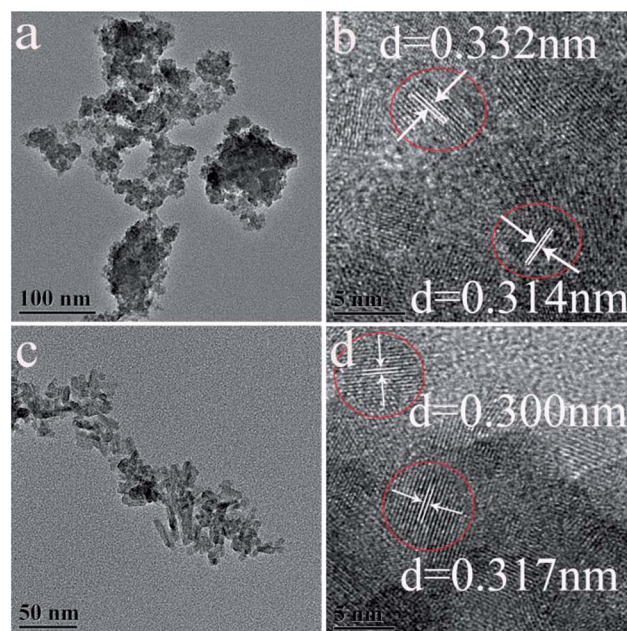


Fig. 2 TEM images of  $\text{LaPO}_4$  nanostructures prepared with citrate-induced (a) the monoclinic phase, (b) HRTEM image of the monoclinic phase, (c) the hexagonal phase, (d) HRTEM image of the hexagonal phase.



the formula of Debye–Scherrer to calculate. A high-resolution TEM image (Fig. 2b) showed that the spacing of the sample between two adjacent horizontal and vertical lattice planes is 3.32 Å and 3.14 Å (Fig. 2b), close to the  $d_{200}$  (3.326 Å) and  $d_{120}$  (3.127 Å), respectively. When citrate concentration was 3.0 mmol, the as-obtained hexagonal phase product has rod-like morphology, and its diameter was 5–10 nm (Fig. 2c). The high-resolution TEM image of  $\text{LaPO}_4$  indicated that the spacing of the sample between two adjacent horizontal and vertical lattice planes is 3.00 Å and 3.17 Å (Fig. 2d), close to the  $d_{200}$  (3.0662 Å) and  $d_{111}$  (3.1057 Å), respectively.

Comparative experiments were carried out to investigate the effects of the ligand citrate on the structure and morphology of the products. The citrate concentration was changed from 0.0 to 3.0 mmol. Fig. 3 showed the XRD pattern of the products synthesized with different citrate concentration. XRD pattern of the product prepared without citrate was shown in Fig. 3a. All the diffraction peaks agree well with hexagonal phase  $\text{LaPO}_4$  (JCPDS 75-1881). When the citrate concentration was 0.2 mmol, the monoclinic phase appeared but the majority phase was hexagonal phase (Fig. 3b). When the citrate concentration was 0.5 mmol, the diffraction intensity of monoclinic phase enhanced (Fig. 3c). While citrate concentration was reaching 0.8 mmol, all diffraction peaks of the product fit well with the diffraction peaks of pure monoclinic phase  $\text{LaPO}_4$  (JCPDS 32-0493), and no hexagonal phase was observed (Fig. 3d). However, when citrate concentration increased to 0.9 mmol, the diffraction peak of the hexagonal phase appeared again (Fig. 3e). With the citrate concentration increasing to 1.0 mmol, the diffraction intensity of the hexagonal phase enhanced gradually (Fig. 3f). When the citrate concentration was reaching to 3.0 mmol, the products transform into a hexagonal phase again (Fig. 3g and h).

The above comparative experiments showed that the citrate-induced the phase transformation of  $\text{LaPO}_4$ . The corresponding SEM images were showed in Fig. 4. Fig. 4a indicated the image of product prepared without citrate. The product was composed

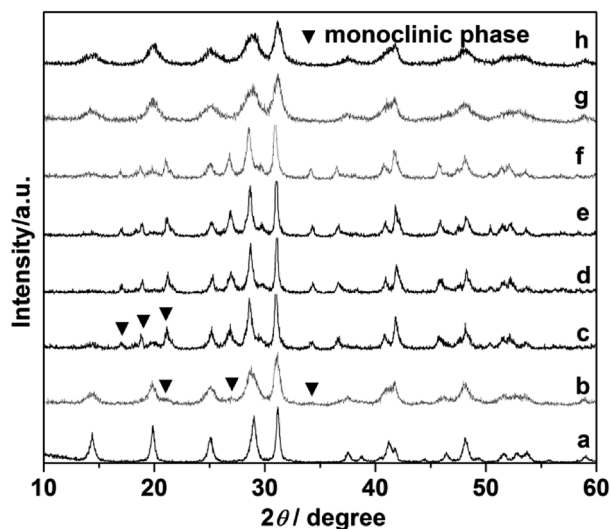


Fig. 3 XRD pattern of the  $\text{LaPO}_4$  prepared with different citrate concentration: (a) 0, (b) 0.2, (c) 0.5, (d) 0.8, (e) 0.9, (f) 1.0, (g) 2.0, (h) 3.0 mmol.

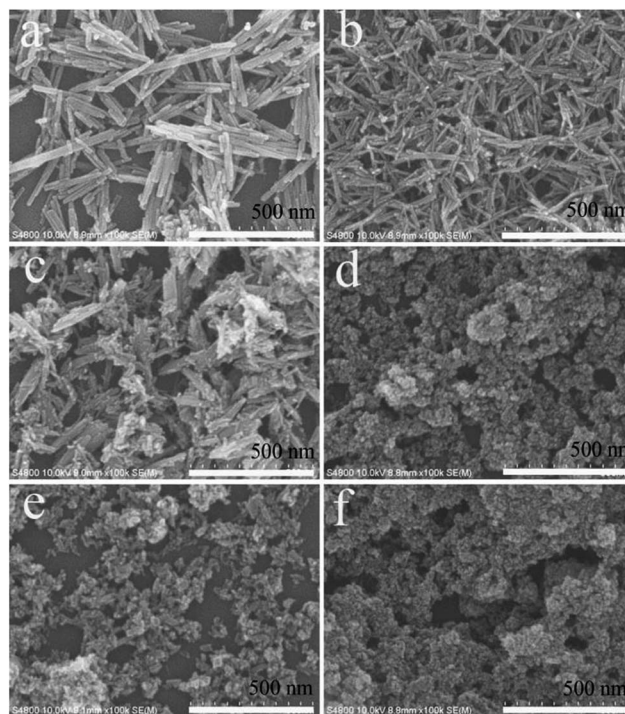


Fig. 4 SEM images of the  $\text{LaPO}_4$  nanostructures prepared with different citrate concentration: (a) 0, (b) 0.2, (c) 0.5, (d) 0.8, (e) 1.0, (f) 3.0 mmol.

with nanowires with a diameter about 20–30 nm and length of 300–500 nm. A large number of uniform nanorods with a length of 80–100 nm and a diameter of 10–20 nm were observed in the product, when the citrate concentration was 0.2 mmol (Fig. 4b). While the citrate concentration was changed from 0.5 to 1.0 mmol, the morphology of the products was rod-like nanoparticles and nanoparticles (Fig. 4c–e). As the citrate concentration increased to 3.0 mmol, the product was composed of nanorods with a diameter of 5–10 nm and length about 20–30 nm (Fig. 4f).

In our reaction system, the monoclinic phase  $\text{LaPO}_4$  could be successfully prepared with citrate-induced at low temperature. When the citrate concentration was 0.8 mmol, the monoclinic phase  $\text{LaPO}_4$  was obtained. Citrate served as ligand and chelating agent of the  $\text{La}^{3+}$  ions, it might result in the growth of initial monoclinic  $\text{LaPO}_4$  particles.<sup>21</sup> However, when the citrate concentration was increased to 0.9 mmol, the mixed hexagonal and monoclinic phase was obtained. The citrate concentration was further increased to 1.5 mmol, the hexagonal phase  $\text{LaPO}_4$  was formed, the phase transformation cannot complete. In addition, we also investigated the effect of citrate concentration on the phase transformation of  $\text{CePO}_4$ . Firstly, we found that the hexagonal phase  $\text{CePO}_4$  was prepared when the reaction temperature was 100 °C and the citrate concentration varying from 0.0 mmol to 3.0 mmol (Fig. S1†). Subsequently, when the reaction temperature was increased to 150 °C and the citrate concentration was increased from 0.0 mmol to 3.0 mmol, XRD patterns of the  $\text{CePO}_4$  was showed in Fig. 5. If the  $\text{CePO}_4$  was prepared without citrate, all the diffraction peaks would agree well with hexagonal  $\text{CePO}_4$  (JCPDS 34-1380). With the citrate concentration increasing to



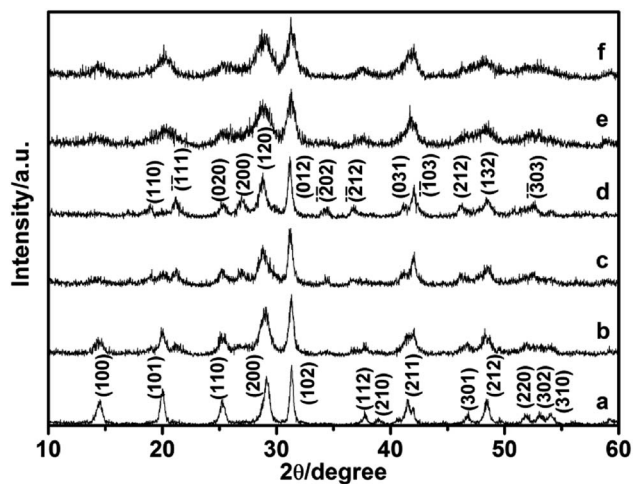


Fig. 5 XRD pattern of the  $\text{CePO}_4$  prepared with different citrate concentration: (a) 0, (b) 0.2, (c) 1.0, (d) 1.2, (e) 2.0, (f) 3.0 mmol.

0.2 mmol, the mixed hexagonal and monoclinic phase appeared (Fig. 5b). Fig. 5c showed that the diffraction of monoclinic phase enhanced gradually while citrate was 1.0 mmol. Furthermore, the citrate concentration increased to 1.2 mmol, all diffraction peaks can be indexed to monoclinic  $\text{CePO}_4$  (JCPDS 32-0199), with no hexagonal phases being observed (Fig. 5d). When citrate increased to 2.0 mmol, diffraction peaks of the mixed hexagonal and monoclinic phase appeared again (Fig. 5e). Up to 3.0 mmol citrate concentration, the product completely transformed into pure hexagonal phase (Fig. 5f). The experimental result showed that the monoclinic phase  $\text{CePO}_4$  prepared with 1.2 mmol citrate-induced at 150 °C. However, if the citrate concentration was higher than 1.2 mmol, the monoclinic phase  $\text{CePO}_4$  cannot be obtained. It further confirms that the citrate concentration have remarkable impact on  $\text{CePO}_4$  transformation. While the citrate concentration was increased, it was possible to increase pH value of the solution. The phase transformation will be influenced by pH value of reaction solution. Therefore, pH values of the reaction solution were further measured at different reaction stages times (Table 1). It can be seen that pH values of reaction solution increased with increasing the citrate concentration. The reaction environment of  $\text{La}^{3+}$  ions and  $\text{PO}_4^{3-}$  began changing when the citrate concentration was higher. When the citrate concentration was 0.8 mmol, pH

(2) value was 1.69, the monoclinic phase  $\text{LaPO}_4$  was synthesized by a citrate-induced at 100 °C. Interestingly, as the citrate concentration was 0.9 mmol, pH (2) value was 1.88, the mixed hexagonal and monoclinic phase  $\text{LaPO}_4$  could be obtained. Then the pure hexagonal phase was synthesized when the citrate concentration was 1.5 mmol and pH (2) value was 3.57. Therefore, pH values of reaction solution increases with increasing the citrate concentration, which disturb the citrate-induced function, and the phase transformation cannot complete.

To further understand the pH values influence on the phase transformation of the products, the comparative experiments were employed using ammonia or nitric acid to adjust the pH value without citrate-induced. If the products prepared without citrate-induced, the hexagonal phase  $\text{LaPO}_4$  could be obtained at 100 °C (Fig. S2a†), and the mixed hexagonal and monoclinic phase was prepared at 150 °C (Fig. S2b†). At 170 °C, the product was monoclinic phase (Fig. S2c†). Subsequently, when the reaction temperature was 100 °C and 170 °C, the products prepared with hydrothermal route using 0.1 mol  $\text{dm}^{-3}$   $\text{H}_3\text{PO}_4$  and  $\text{La}(\text{NO}_3)_3$  as reactant using ammonia or nitric acid to adjust the pH value of the reaction solution. Fig. S3 and S4† showed XRD pattern of  $\text{LaPO}_4$  obtained at 100 °C and 170 °C with different pH value. When the reaction temperature was 100 °C, and pH values were 1.0, 5.0, 7.0, and 9.0, all diffraction peaks of the products coincided with hexagonal phase (Fig. S3†). With the reaction temperature increasing to 170 °C and pH = 1.0, the monoclinic phase  $\text{LaPO}_4$  was synthesized (Fig. S4a†). However, if the pH values of reaction system were greater than 2.0, all products would exhibit hexagonal phase structure (Fig. S4b-e†). The monoclinic phase  $\text{LaPO}_4$  could be synthesized without citrate-induced at 170 °C, but pH value of the reaction system should be smaller than 2.0. The pH value would be higher than 1.69 when the concentration of citrate was greater than 0.8 mmol. Furthermore, We found that the citrate-induced function would become unavailable under this condition. That is, the pure monoclinic phase  $\text{LaPO}_4$  cannot be obtained and the hexagonal phase  $\text{LaPO}_4$  was synthesized.

Therefore, we suggest a possible formation mechanism of the influence of citrate on phase transformation based on the above results (Fig. 6). First, when the citrate concentration increased to 0.8 mmol, the hexagonal phase  $\text{LaPO}_4$  transformed into monoclinic phase. During the process of synthesis, pH

Table 1 pH values of the reaction system, when the  $\text{LaPO}_4$  synthesized with different concentration of citrate<sup>a</sup>

Samples	$\text{La}^{3+}(\text{mol L}^{-1})$	Citrate (mmol)	pH (1)	pH (2)	Samples structure
SM1	1.0	0.0	4.92	1.08	Hexagonal phase
SM2	1.0	0.2	4.47	1.14	Mixed hexagonal and monoclinic phase
SM3	1.0	0.5	4.25	1.34	Mixed hexagonal and monoclinic phase
SM4	1.0	0.8	4.28	1.69	Monoclinic phase
SM5	1.0	0.9	4.36	1.88	Mixed hexagonal and monoclinic phase
SM6	1.0	1.0	4.70	2.20	Mixed hexagonal and monoclinic phase
SM7	1.0	1.2	4.93	2.32	Mixed hexagonal and monoclinic phase
SM8	1.0	1.5	5.82	3.57	Hexagonal phase
SM9	1.0	2.0	6.25	4.24	Hexagonal phase

<sup>a</sup> pH (1) was the pH value of the reaction solution after citrate added to  $\text{La}(\text{NO}_3)_3$  solution, and pH (2) was the pH value of the reaction solution after added  $\text{H}_3\text{PO}_4$ .



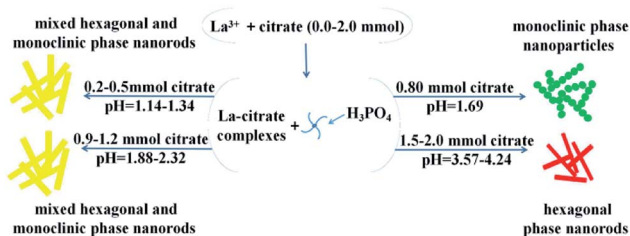


Fig. 6 The crystal growth mechanism of the  $\text{LaPO}_4$  nanostructures with hexagonal and monoclinic phase.

value was 1.69. The citrate possesses three carboxylic functional groups, which chelated with  $\text{RE}^{3+}$  ions to form RE-citrate complex. Then RE-citrate complex reacted with  $\text{PO}_4^{3-}$  to produce  $\text{REPO}_4$  via the substitution reaction after adding the phosphorus source  $\text{H}_3\text{PO}_4$ . In the reaction procedure, the presence of citrate complexes caused the  $\text{RE}^{3+}$  to be slowly released.<sup>21</sup> It is quite possible that the substituted citrate ions are adsorbed onto the surface of the initially formed tiny hexagonal  $\text{REPO}_4$  nanoparticles around  $\text{RE}^{3+}$  cation, which might result in the growth of initial monoclinic  $\text{REPO}_4$  particles.<sup>20</sup> Second, if the citrate concentration increasing higher than 0.8 (pH > 1.69), the condition would not conducive to the preferential growth of monoclinic phase nanoparticles. Third, with the citrate concentration increased from 0.9 mmol to 1.2 mmol (pH = 1.88–2.32), the mixed hexagonal phase and monoclinic phase  $\text{LaPO}_4$  was obtained. Finally, when the citrate concentration was higher than 1.5 mmol (pH > 3.57), the hexagonal phase  $\text{LaPO}_4$  was synthesized but the monoclinic phase  $\text{LaPO}_4$  could not be obtained.

### 3.2 Photoluminescence properties

5%  $\text{Eu}^{3+}$ -doped  $\text{LaPO}_4$  with different phase prepared by citrate-induced. We found that the addition of 5%  $\text{Eu}^{3+}$  to the raw  $\text{La}^{3+}$  solution did not affect the crystalline structure (Fig. S5†). EDX analysis clearly indicated the presence of specific dopant  $\text{Eu}^{3+}$  ion in the synthesized product (Fig. S6†). It can be confirmed by the content of the elements shown in EDX that the synthesized product was  $\text{La}_{0.95}\text{PO}_4:\text{Eu}_{0.05}$ .

The photoluminescence (PL) spectra were measured at room temperature. The excitation peak of the products centered at 393 nm (Fig. S7†). Fig. 7 exhibits the emission spectra of  $\text{LaPO}_4:\text{Eu}$  with different phase under excitation at 393 nm. The emission spectra exhibits four characteristic emission lines, which were attributed to the  ${}^5\text{D}_0 \rightarrow {}^7\text{F}_j$  ( $J = 1-4$ ) transitions of  $\text{Eu}^{3+}$ . The magnetic dipole transition ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$ ) occupied a dominate position in as-synthesized hexagonal phase (Fig. 7a and c). It is known that while the  $\text{Eu}^{3+}$  is in the symmetry center of the lattice, the electric dipole transition is forbidden, the intensity of the  ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$  transition band in the emission spectra is weak, the intensity of the  ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$  transition band is strong.<sup>26,27</sup> In addition, the emission peaks split to multi-peaks. It was due to that the perturbation of the crystal field and the change of the local site symmetry, the degeneracy of  ${}^7\text{F}_j$  ( $J = 1-4$ ) energy level was resolved.<sup>28</sup> However, the monoclinic phase prepared with citrate-induced has higher intensity at the

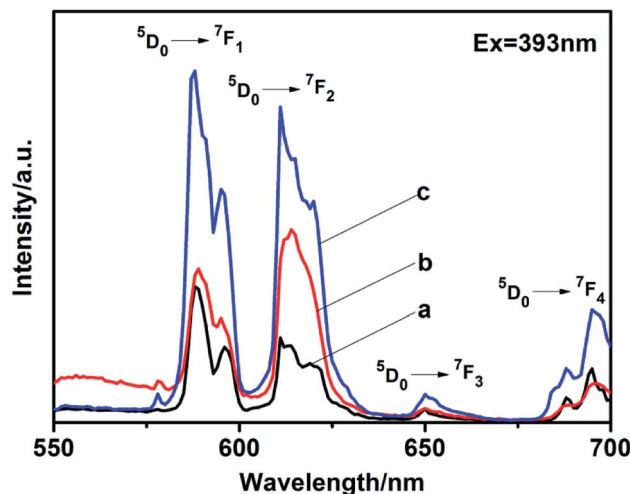


Fig. 7 Emission spectra of  $\text{LaPO}_4:\text{Eu}$  with different phase: (a) hexagonal phase prepared without citrate, (b) monoclinic phase prepared with citrate-induced, (c) hexagonal phase prepared with citrate-induced.

electric dipole transition ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$ ) than that at the magnetic dipole transition ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$ ), which means  $\text{Eu}^{3+}$  is in a site with no inversion center (Fig. 7b).<sup>29-31</sup> While the absolute quantum yield of the monoclinic phase  $\text{LaPO}_4:\text{Eu}$  is 10.71%. Moreover, the emission intensity of the hexagonal phase prepared with citrate-induced is stronger than that of the hexagonal phase prepared without citrate. The absolute quantum yield of the hexagonal phase  $\text{LaPO}_4:\text{Eu}$  prepared with citrate-induced is 14.73%, while the hexagonal phase  $\text{LaPO}_4:\text{Eu}$  prepared without citrate is 9.24%. The size, morphology and structure may also influenced the photoluminescence intensities.<sup>32-35</sup> The photoluminescence fitting curves of  $\text{LaPO}_4:\text{Eu}$  with different phase were also recorded, as shown in Fig. S8†. The calculated average lifetime ( $\tau$ ) was 0.82 ms for the monoclinic phase  $\text{LaPO}_4:\text{Eu}$  prepared with citrate-induced. Furthermore, the calculated average lifetimes ( $\tau$ ) were 1.83 ms and 1.51 ms for the hexagonal phase  $\text{LaPO}_4:\text{Eu}$  prepared with citrate-induced and without citrate, respectively.

## 4 Conclusions

In this paper, we have developed a low temperature and citrate-induced hydrothermal route for the crystal growth of monoclinic phase  $\text{LaPO}_4$  and  $\text{LaPO}_4:\text{Eu}$  nanostructures. The conclusions of the research were as follows: (1) when the monoclinic phase  $\text{LaPO}_4$  was synthesized through hydrothermal route at 170 °C, the pH value of reaction system should be less than or equal to 1.0. (2) The monoclinic phase  $\text{LaPO}_4$  was controlled synthesized through citrate-induced at 100 °C, when the concentration of citrate was 0.8 mmol. High concentration of citrate would result in the pH value of the reaction system higher than 1.88, which cannot synthesize the monoclinic phase  $\text{LaPO}_4$  with citrate-induced. The hexagonal phase  $\text{LaPO}_4$  was obtained. (3) Significantly, the photoluminescence properties demonstrate that the electric dipole transition ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$ ) is stronger than the magnetic dipole transition ( ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$ ) for



the monoclinic phase  $\text{LaPO}_4:\text{Eu}$  prepared with citrate-induced. The hexagonal phase  $\text{LaPO}_4:\text{Eu}$  with citrate-induced have stronger emission intensity than the hexagonal phase  $\text{LaPO}_4:\text{Eu}$  without citrate-induced. Due to low temperature synthesis method of this system and the effective control over the phase transformation it has achieved, we suppose this study may have broad application prospects in exploring crystal growth process.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- 1 S. Heer, O. Lehmann, M. Haase and H. U. Güdel, *Angew. Chem., Int. Ed.*, 2003, **42**, 3179–3182.
- 2 A. I. Becerro, S. Rodríguez-Liviano, A. J. Fernández-Carrión and M. Ocaña, *Cryst. Growth Des.*, 2013, **13**, 526–535.
- 3 G. F. Ju, Y. H. Hu, L. Chen, X. J. Wang, Z. F. Mu, H. Y. Wu and F. W. Kang, *J. Alloys Compd.*, 2011, **509**, 5655–5659.
- 4 K. Park and M. H. Heo, *J. Alloys Compd.*, 2011, **509**, 9111–9115.
- 5 S. H. Lee, K. Teshima, S. Mori, M. Endo and S. J. Oishi, *Cryst. Growth Des.*, 2010, **10**, 1693–1698.
- 6 B. Sun and H. Sirringhaus, *Nano Lett.*, 2005, **5**, 2408–2413.
- 7 M. F. Dumont, C. Baligand, Y. C. Li, E. S. Knowles, M. W. Meisel, G. A. Walter and D. R. Talham, *Bioconjugate Chem.*, 2012, **23**, 951–957.
- 8 R. X. Yan, X. M. Sun, X. Wang, Q. Peng and Y. D. Li, *Chem.–Eur. J.*, 2005, **11**, 2183–2195.
- 9 Y. P. Fang, A. W. Xu, R. Q. Song, H. X. Zhang, L. P. You, J. C. Yu and H. Q. Liu, *J. Am. Chem. Soc.*, 2003, **125**, 16025–16034.
- 10 K. S. Gavrichev, M. A. Ryumin, A. V. Tyurin, A. V. Khoroshilov, L. P. Mezentseva, A. V. Osipov, V. L. Ugolkov and V. V. Gusarov, *J. Therm. Anal. Calorim.*, 2010, **102**, 809–811.
- 11 Y. J. Zhang and H. M. Guan, *J. Cryst. Growth*, 2003, **256**, 156–161.
- 12 S. Z. Lin, X. T. Dong, R. K. Jia and Y. L. Yuan, *J. Mater. Sci.: Mater. Electron.*, 2010, **21**, 38–44.
- 13 C. R. Kesavulu, C. Basavapoornima, C. S. Dwaraka Viswanath and C. K. Jayasankar, *J. Lumin.*, 2016, **171**, 51–57.
- 14 K. Riwozki, H. Meyssamy, A. Kornowski and M. Haase, *J. Phys. Chem. B*, 2000, **104**, 2824–2828.
- 15 T. Huang, S. R. Shieh, A. Akhmetov, X. Liu, C. M. Lin and J. S. Lee, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 214117.
- 16 D. Errandonea, J. Pellicer-Porres, F. J. Manjón, A. Segura, Ch. Ferrer-Roca, R. S. Kumar, O. Tschauner, P. Rodríguez-Hernández, J. López-Solano, S. Radescu, A. Mujica, A. Muñoz and G. Aquilanti, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 174106.
- 17 J. M. Heuser, R. I. Palomares, J. D. Bauer, M. J. Lozano Rodriguez, J. Cooper, M. Lang, A. C. Scheinost, H. Schlenz, B. Winkler, D. Bosbach, S. Neumeier and G. Deissmann, *J. Eur. Ceram. Soc.*, 2018, **38**, 4070–4081.
- 18 M. A. Musselman, T. M. Wilkinson, B. Haberl and C. E. Packard, *J. Am. Ceram. Soc.*, 2018, **101**, 2562–2570.
- 19 F. Luo, C. J. Jia, W. Song, L. P. You and C. H. Yan, *Cryst. Growth Des.*, 2005, **5**, 137–142.
- 20 J. R. Bao, R. B. Yu, J. Y. Zhang, D. Wang, J. X. Deng, J. Chen and X. R. Xing, *Scr. Mater.*, 2010, **62**, 133–136.
- 21 N. O. Nuñez, S. R. Liviano and M. Ocaña, *J. Colloid Interface Sci.*, 2010, **349**, 484–491.
- 22 J. Q. Hu, Q. Chen, Z. X. Xie, G. B. Han, R. H. Wang, B. Ren, Y. Zhang, Z. L. Yang and Z. Q. Tian, *Adv. Funct. Mater.*, 2004, **14**, 183–189.
- 23 X. H. Ji, X. N. Song, J. Li, Y. B. Bai, W. S. Yang and X. G. Peng, *J. Am. Chem. Soc.*, 2007, **129**, 13939–13948.
- 24 W. H. Di, M. G. Willinger, R. A. S. Ferreira, X. G. Ren, S. Z. Lu and N. Pinna, *J. Phys. Chem. C*, 2008, **112**, 18815–18820.
- 25 L. Zhang, L. L. Fu, X. X. Yang, Z. L. Fu, X. D. Qi and Z. J. Wu, *J. Mater. Chem. C*, 2014, **2**, 9149–9158.
- 26 Y. G. Yang, *Mater. Sci. Eng., B*, 2013, **178**, 807–810.
- 27 C. Zollfrank, H. Scheel, S. Brungs and P. Greil, *Cryst. Growth Des.*, 2008, **8**, 766–770.
- 28 Y. H. Zheng, H. P. You, G. Jia, K. Liu, Y. H. Song, M. Yang and H. J. Zhang, *Cryst. Growth Des.*, 2009, **9**, 5101–5107.
- 29 X. Y. Huang, H. Guo and B. Li, *J. Alloys Compd.*, 2017, **720**, 29–38.
- 30 X. Y. Huang, B. Li, H. Guo and D. Q. Chen, *Dyes Pigm.*, 2017, **143**, 86–94.
- 31 P. Du, X. Y. Huang and J. S. Yu, *Chem. Eng. J.*, 2018, **337**, 91–100.
- 32 M. Wang, Q. L. Huang, J. M. Hong, X. T. Chen and Z. L. Xue, *Cryst. Growth Des.*, 2006, **6**, 1972–1974.
- 33 J. Geng, Y. N. Lv, D. J. Lu and J. J. Zhu, *Nanotechnology*, 2006, **17**, 2614–2620.
- 34 X. C. Wu, Y. R. Tao, C. Y. Song, C. J. Mao, L. Dong and J. J. Zhu, *J. Phys. Chem. B*, 2006, **110**, 15791–15796.
- 35 L. M. Chen, Y. N. Liu and K. L. Huang, *Mater. Res. Bull.*, 2006, **41**, 158–166.

