

Cite this: *Chem. Sci.*, 2018, 9, 6779

All publication charges for this article have been paid for by the Royal Society of Chemistry

Received 6th June 2018
Accepted 29th June 2018

DOI: 10.1039/c8sc02492h

rsc.li/chemical-science

Interfacing porphyrins and carbon nanotubes through mechanical links†

Leire de Juan-Fernández,^{‡a} Peter W. Münich,^{‡b} Arjun Puthiyedath,^{id b} Belén Nieto-Ortega,^{id a} Santiago Casado,^a Luisa Ruiz-González,^c Emilio M. Pérez^{id *a} and Dirk M. Guldi^{id *b}

We describe the synthesis of rotaxane-type species composed of macrocyclic porphyrin rings mechanically interlocked with SWCNT threads. The formation of mechanically interlocked SWCNTs (MINTs) proceeds with chiral selectivity, and was confirmed by spectroscopic and analytical techniques and adequate control experiments, and corroborated by high-resolution electron microscopy. From a thorough characterization of the MINTs through UV-vis-NIR absorption, fluorescence, Raman, and transient absorption spectroscopy we analyse in detail the electronic interactions of the porphyrins and the SWCNTs in the ground and excited states.

Introduction

The ability of metalloporphyrins to reversibly coordinate gases is exploited by nature in the transport of oxygen by the heme group. Directly related to this is another main biological function of porphyrinoid structures, that is, as co-factors in cytochrome P450, which is responsible for many catalytic redox reactions. Finally, the intense absorption in the visible region and the ability to accept electrons upon photoexcitation of the structurally related chlorophylls renders them ideal light antennas and primary electron acceptors in photosynthesis. Inspired by these crucial roles, porphyrins are one of the most extensively studied organic chromophores.^{1–5} A particularly active area of research is the combination of porphyrins and carbon nanotubes (CNT), which also benefits from the extraordinary physical properties of CNTs.^{6–9} Porphyrin–CNT hybrids have been successfully applied to the fabrication of bioinspired high-performance gas sensors,^{10,11} catalysts,^{12–15} model photosynthetic systems,^{16–18} and other optoelectronic devices.¹⁹ The physical properties of all these artificial systems depend crucially on the nature of the linker between them.

A covalent porphyrin–CNT connection typically leads to good kinetic stability, and the structure of the spacer can be used to

modulate porphyrin–CNT communication.^{20–22} Interfacing porphyrins and CNTs *via* non-covalent interactions provides an alternative means to generate porphyrin–CNT hybrids.^{23–29} The intrinsic porphyrin–CNT non-covalent interactions have been skillfully exploited to associate and purify single-walled CNTs (SWCNT) in a chirality and enantioselective fashion.^{30–34}

Mechanically interlocked molecules (MIM) are composed of covalent components linked together by their topology. Rotaxanes are prototypical examples of MIMs, in which one or more macrocycles are linked to a dumbbell-shaped linear component: the thread. The mechanical bond imparts MIMs with unique dynamic properties.^{35,36} Specifically, the macrocycles can be moved along or around the thread through controlled submolecular motion. This is a key step for the construction of some of the most advanced examples of synthetic molecular machines.^{37–39}

We have developed methods to thread SWCNTs through macrocycles to form rotaxane-like mechanically interlocked nanotube derivatives (MINT) using a clipping strategy.⁴⁰ More recently, groups led by Kruss,⁴¹ and Miki and Ohe,⁴² have described the encapsulation of SWCNTs into peptide barrels and rigid cycloparaphenyleneacetylenes, respectively. The native structure of SWCNTs is preserved upon formation of MINTs, while the extreme aspect ratio of SWCNTs prevent the macrocycles from dissociating, providing MINTs with a kinetic stability comparable to that of covalently functionalized SWCNTs.^{43–45} We have also documented that MINTs feature unique physical properties, which are quite different from classic supramolecular SWCNT derivatives.^{46,47}

Here, we interface porphyrins and SWCNTs through mechanical links. The MINT-forming reaction proceeds with chiral selectivity for the smaller diameter SWCNTs. The electronic interactions between the porphyrins and the nanotubes

^aIMDEA Nanoscience, C/ Faraday 9, Ciudad Universitaria de Cantoblanco, 28049 Madrid, Spain. E-mail: emilio.perez@imdea.org

^bDepartment of Chemistry and Pharmacy, Interdisciplinary Center for Molecular Materials, Friedrich-Alexander University of Erlangen-Nürnberg, Egerlandstr. 3, 91058 Erlangen, Germany. E-mail: guldi@fau.de

^cDepartamento de Química Inorgánica, Fac. C. C. Químicas, Universidad Complutense de Madrid, Avenida Complutense s/n, 28040 Madrid, Spain

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8sc02492h

‡ These authors contributed equally.

in the ground state, as well as the charge separation and recombination processes and their underlying dynamics upon photoexcitation are also described in detail.

Results and discussion

Following experimental fine-tuning (see the ESI†) we focused on a U-shaped molecule equipped with two porphyrin recognition units for SWCNTs (**U-por** in Fig. 1). A hexamethylene spacer provides flexibility to template the macrocyclization around SWCNTs through ring-closing metathesis (RCM) of the terminal alkenes, to form the corresponding MINTs. The synthesis and full characterization of **U-por** and **mac-por** is described in the ESI.†

We explored the formation of **MINT-por** through templated RCM of **U-por** in the presence of (6,5)-enriched SWCNTs. The diameter of the latter is 0.75 nm, which is a good fit for the cavity of **mac-por**, as shown by the energy-minimized molecular structure (HF-3c)⁴⁸ shown in Fig. 1B. SWCNTs (2 mg) were suspended in tetrachloroethane (TCE, 2 mL) through sonication and mixed with **U-por** (4.2 mM) and Grubbs' 2nd generation catalyst at room temperature (rt) for 72 hours.

Next, the suspension was filtered through a polytetrafluoroethylene membrane with a pore size of 0.2 μm and washed with dichloromethane (DCM) to remove non-threaded macrocycles, catalyst and any remaining linear precursors. Three resuspension–filtration iterations complete the purification. Hereafter, the samples were dried and subjected to thermogravimetric analysis (TGA) to quantify the degree of functionalization. TGA showed a weight loss of 24% at around 340 $^{\circ}\text{C}$, corresponding to the porphyrinic material. We performed experiments in which the concentration of **U-por** was varied between 1.06 and 8.4 mM to modulate the degree of functionalization. The product formed at the lowest

concentration of the U-shaped precursor showed a loss of 10% at 340 $^{\circ}\text{C}$, whereas a weight loss of 26% was observed for the highest concentration (see the ESI†).⁴⁹ In addition, the derivative of the TGA data features a single peak for the porphyrinic material, confirming that oligomers of **U-por** adsorbed onto SWCNTs are absent in **MINT-por**.^{43,50,51}

To test the stability of **MINT-por**, we heated the samples for 30 min under reflux in TCE (bp = 147 $^{\circ}\text{C}$), followed by a thorough rinse with DCM. TGA of the resulting samples showed a subtle decrease of 3% in functionalization confirming the MINT stability.

Investigations of **MINT-por** by means of atomic force microscopy (AFM) and transmission electron microscopy (TEM) corroborate the presence of rotaxane-type species. Fig. 2A shows a typical AFM topographic image of a spin-casted TCE suspension of **MINT-por**. Individualized SWCNTs with heights around 0.8 nm decorated with objects, which are consistently around 2.0 nm high, are in sound agreement with **mac-por** around a SWCNT (Fig. 1B). Macrocycles adsorbed onto SWCNTs rather

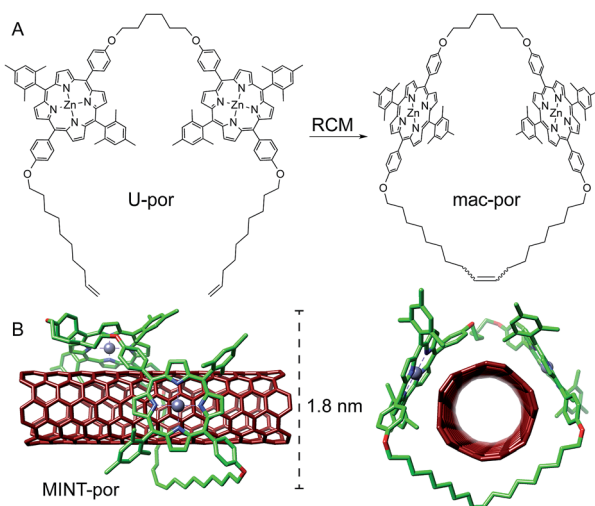


Fig. 1 (A) Chemical structures of **U-por** and **mac-por**. (B) Side and front view of the energy-minimized structure (HF-3c) of **MINT-por**, showing the diameter of the macrocycle. C atoms in green (**mac-por**) or dark red (SWCNT), N in blue, O in red, Zn in metallic gray. H atoms omitted for clarity.

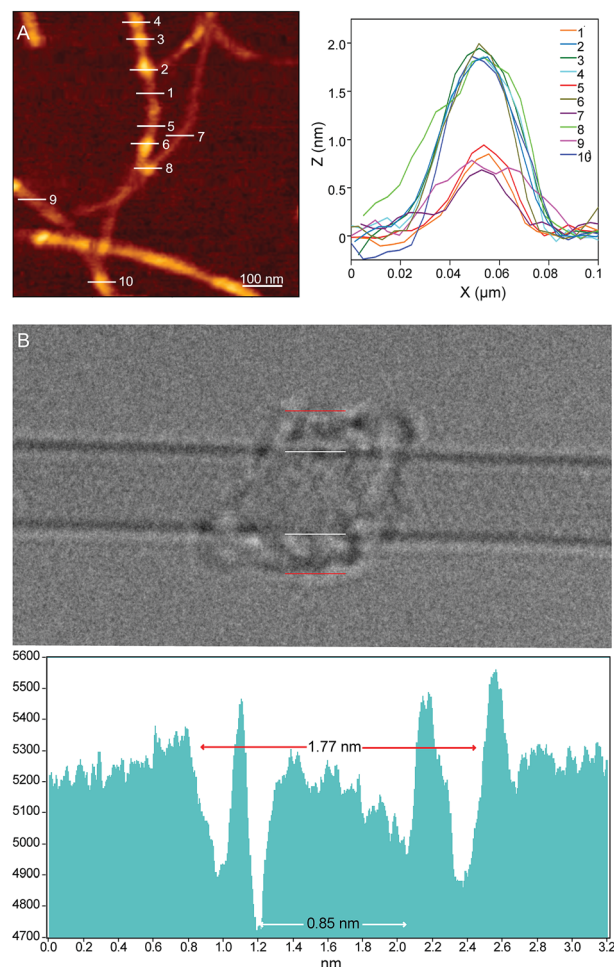


Fig. 2 (A) AFM topographic image of a suspension of **MINT-por** in TCE and corresponding height profiles along the lines marked in the topographic image. (B) AC-HRTEM image of an individual SWCNT functionalized by a single macrocycle in **MINT-por** and corresponding profiles (white lines and arrows mark SWCNT diameter, red lines and arrows mark **mac-por** diameter).



than threaded by SWCNTs are expected to show a significantly smaller height.

Aberration-corrected high resolution TEM (AC-HRTEM) images further confirm the rotaxane-like structure of **MINT-por**. They were drop-casted from an isopropanol suspension onto copper grids and reveal mostly bundled SWCNTs with functionalized sidewalls. These co-exist, however, with isolated SWCNTs of 0.8 nm diameter, in which single circular objects of a size of *ca.* 1.8 nm are discernable. A representative AC-HRTEM micrograph of **MINT-por** with close to atomic resolution is shown in Fig. 2B, which shows a SWCNT with a diameter of 0.85 nm, surrounded by a macrocycle that shows a diameter of 1.77 nm. In contrast to our previous MINTs, in which we observed macrocycle-like rings around SWCNTs,¹³ the large dimensions of the porphyrins – in the range of 1.4–1.8 nm, comparable to the diameter of the macrocycle – confer the macrocycles a box-like shape. We note that the distances between the macrocycle and the SWCNT sidewalls correspond to very close van der Waals contacts (<0.4 nm). Finally, from the difference in contrast between the SWCNT walls we conclude that **mac-por** is threaded by SWCNTs rather than physisorbing on them. We note that, even under these low voltage conditions (see the ESI[†]), the sample is eventually damaged under the e-beam, most likely starting with the more sensitive C–H bonds of the alkyl chain,⁵² but eventually damaging even the SWCNT wall (see Fig. S2[†]).

To gain insights into the electronic properties of **MINT-por**, we carried out UV-vis-NIR, fluorescence, and Raman spectroscopy measurements. Steady state absorption spectra (D₂O/SDBS (1 wt%, rt)), of (6,5)-enriched SWCNTs (black) and the corresponding MINTs (red) are compared in Fig. 3A. In the **MINT-por** sample, the Soret-band absorption of the porphyrins is evident at 430 nm. As far as the absorption spectra of (6,5)-enriched SWCNTs are concerned, S₂₂ transitions are noted in the visible region, while the S₁₁ transitions dominate the nIR region. In particular, (6,5)-SWCNT related maxima appear at 570 and 981 nm. Features of (7,5)-SWCNTs are seen at 649 and 1022 nm, whereas those of (7,6)-SWCNTs evolve at 664 and 1136 nm. In **MINT-por**, all of the aforementioned features are red-shifted to 572, 982, 652, 1022, 664, and 1137 nm, on one hand, and broadened, on the other hand. From the aforementioned we conclude weak electronic interactions between SWCNTs and the porphyrin in the ground state. When turning to the porphyrin absorption spectrum, the Soret-band absorption maximizes at 429 nm in DMF for **MINT-por** as well as for **U-por** – Fig. S3.[†]⁵³

To probe the chirality selectivity during the MINT formation,⁴⁷ the relative absorption intensities in dispersions of pristine SWCNTs and **MINT-por** are inspected. A look at, for example, the (6,5)-SWCNTs features in the nIR region reveals that the relative intensity is higher for SWCNTs than for **MINT-por**. The trend is reversed for (7,5)-SWCNTs. Implicit is a chiral selectivity of the MINT formation based on different SWCNT diameters.

To shed light on the excited state interactions in **MINT-por**, the SWCNT fluorescence was analyzed in the nIR-region – Fig. 3B and C. In spectra of SWCNT and **MINT-por**,



Fig. 3 (A) Normalized absorption spectra of (6,5)-enriched SWCNTs (black), and **MINT-por** (red) in D₂O/SDBS (1 wt%, rt), baseline-corrected and normalized to the (7,6)-SWCNT absorption at 1136 nm. 3D nIR fluorescence spectra of (B) (6,5)-enriched SWCNTs and (C) **MINT-por** in D₂O/SDBS (1 wt%, rt) with OD of 0.21 at 570 nm.

characteristic fluorescence features of (8,3), (6,5), (7,5), (8,4) and (7,6)-SWCNTs are present. At first glance, the fluorescence features of pristine SWCNTs at 967, 988, 1031, 1127, and 1130 nm are – in line with the steady state absorption measurements – red-shifted in **MINT-por**; the new maxima are at 967, 993, 1034, 1130, and 1132 nm. A closer look at the fluorescence spectra shown in Fig. S4[†] reveals quenching, which is stronger for small diameter SWCNTs, that is, (8,3)-, (6,5)-, and (7,5)-SWCNT, than for larger diameter SWCNTs, that

is, (8,4)- and (7,6)-SWCNT. A possible rationale is a higher degree of functionalization in the case of smaller diameter SWCNTs in comparison to their larger diameter analogues. Quenching, although moderate in nature, was also observed for the porphyrin fluorescence – Fig. S4.† Quenching in **MINT-por** is likely to stem from deactivation pathways such as energy or electron transfer, which compete with the fluorescent decay.

Electronic interactions as seen in the excited state might also impact the Raman features. To this end, about 1000 Raman spectra were taken for each sample – Fig. 4. They show small up-shifts of the D-, G- and 2D-modes from 1298, 1584 and

2589 cm^{-1} for (6,5)-enriched SWCNTs to 1302, 1590 and 2595 cm^{-1} for **MINT-por**. For **MINT-por** the up-shifts indicate weak charge-transfer interactions, which relate to a donating of charge density from the porphyrin to the SWCNTs.

Raman spectroscopy is also a powerful tool to investigate chiral selectivity. To this end, the focus was on the low frequency radial breathing modes, which are unique for each SWCNT chirality (see Fig. S5†). Here, signals appear at about 255, 282 and 303 cm^{-1} , which stem from (7,6)-, (7,5)-, and (6,5)-SWCNTs, respectively. Well in line with the results from absorption and fluorescence measurements, the relative intensity of (7,5)-SWCNTs is higher in **MINT-por** than in the pristine sample of SWCNTs. Important is the fact that no significant changes were noted in the I_D/I_G ratios when comparing (6,5)-enriched SWCNTs (0.12) with **MINT-por** (0.14). This confirms the absence of defect introduction during the functionalization.

Notable quenching in the fluorescence measurements prompted us to perform femtosecond transient absorption measurements – Fig. 5 and S6.† At first glance, the spectra of (6,5)-enriched SWCNTs and **MINT-por** are dominated by an immediate ground state bleaching of SWCNTs in the nIR-region. Hereby, the nIR bleachings mirror image the steady state absorption spectra. In stark contrast, contributions from the porphyrins in **MINT-por** in the visible region are not discernable as they are masked by the much stronger SWCNT



Fig. 4 Raman histograms of (A) D-mode, (B) G-mode, and (C) 2D-mode of drop-casted SWCNTs (black) and **MINT-por** (red) from methanol, with 633 nm laser excitation.



Fig. 5 Differential absorption spectra of (A) (6,5)-enriched SWCNTs and (B) **MINT-por** obtained upon femtosecond pump probe experiments following 420 nm laser excitation in D_2O /SDBS (1 wt%, rt) at several time delays between 0 and 300 ps.



features. A closer look reveals that for (6,5)-enriched SWCNTs minima evolve at 568, 648, 667, 983, and 1121 nm, while maxima appear at 484, 531, 612, 927, 1073, 1211, and >1300 nm. In **MINT-por**, the same features are discernable but red-shifted to 569, 650, 669, 986 and 1121 nm in the case of the minima and to 490, 535, 617, 927, 1076, 1215 and >1300 nm in the case of the maxima. A three exponential fitting procedure of the ground state recovery of (6,5)-, (7,5)-, (8,4)- and (7,6)-SWCNTs yielded three different lifetimes, which differ for each SWCNT chirality – Table S1.† In SWCNTs, the long-lived component relates to radiative exciton recombination, while the two short-lived components stem from interband- and intertube relaxations. In **MINT-por**, all lifetimes are markedly shorter. Interestingly, smaller diameter (6,5)- and (7,5)-SWCNTs, which are subjective to the strongest fluorescence quenching, give rise to the fastest recovery. The long-lived component is decreased in, for example, (6,5)-SWCNTs by 32% and by 52% for (7,5)-SWCNTs. In stark contrast, in larger diameter (8,4)- and (7,6)-SWCNTs the decrease of the long-lived component is only 15% and 12%, respectively.⁵⁴ We conclude that charge separation in **MINT-por** with 5 ps for (6,5)-SWCNT and 4 ps for (7,5)-SWCNT is followed by charge recombination with 78 and 81 ps, respectively. In the case of larger diameter SWCNTs both charge separation and recombination are less favored: the corresponding lifetimes are 4 and 72 ps for (8,4)- as well as 3 and 52 ps for (7,6)-SWCNTs.

Conclusions

In the current study, we have documented the interfacing of porphyrin chromophores and SWCNTs *via* mechanical interlocking, through the synthesis of rotaxane-type architectures. Formation of MINTs is confirmed *via* adequate control experiments, analytical, spectroscopic and microscopic techniques. Besides TGA, the threading of the macrocycles with SWCNTs to form MINTs without introducing SWCNT defects was confirmed by means of AFM, AC-HRTEM and Raman spectroscopy. Moreover, from Raman and steady state absorption spectroscopy we concluded the selective functionalization of smaller diameter SWCNTs. In line with chiral selectivity, electronic interactions, which are intense between the porphyrin rings and SWCNTs of smaller diameters, are probably the most important factor. Transient absorption spectroscopy proved also to be very important to detect charge separation and recombination in **MINT-por** and to dissect the underlying dynamics. We are currently exploring the consequences that this entirely novel approach to connect porphyrins and SWCNTs has on applications such as catalysis, magnetism, and sensing.^{55,56}

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

Funding from the European Research Council (307609-StG), MINECO (CTQ2014-60541-P, FPU13/03371, JdC-2015-23531, CTQ2017-86060-P) and the Comunidad de Madrid (MAD2D-

CM S2013/MIT-3007) is gratefully acknowledged. IMDEA Nanociencia acknowledges support from the “Severo Ochoa” Programme for Centres of Excellence in R&D (MINECO, Grant SEV-2016-0686). The computational work was supported by the Campus de Excelencia Internacional UAM+CSIC. We thank the Supercomputing and Bioinnovation Center (SCBI) of the University of Málaga (Spain) for their support and resources. We thank the National Centre for Electron Microscopy (ICTS-CNME, Universidad Complutense) for electron microscopy facilities. P. W. M. acknowledges the Graduate School of Molecular Science for funding. This work was further supported by the Deutsche Forschungsgemeinschaft as part of the Excellence Cluster Engineering of Advanced Materials and SFB 953 Synthetic Carbon Allotropes as well as by the Bavarian State Government as part of the Solar Technologies go Hybrid initiative.

Notes and references

- 1 L.-L. Li and E. W.-G. Diau, *Chem. Soc. Rev.*, 2013, **42**, 291–304.
- 2 M. Ethirajan, Y. Chen, P. Joshi and R. K. Pandey, *Chem. Soc. Rev.*, 2011, **40**, 340–362.
- 3 J. A. A. W. Elemans, R. van Hameren, R. J. M. Nolte and A. E. Rowan, *Adv. Mater.*, 2006, **18**, 1251–1266.
- 4 J. L. Sessler and D. Seidel, *Angew. Chem., Int. Ed.*, 2003, **42**, 5134–5175.
- 5 D. M. Guldi, *Chem. Soc. Rev.*, 2002, **31**, 22–36.
- 6 T. Umeyama and H. Imahori, *J. Phys. Chem. C*, 2013, **117**, 3195–3209.
- 7 D. M. Guldi and R. D. Costa, *J. Phys. Chem. Lett.*, 2013, **4**, 1489–1501.
- 8 M. F. L. De Volder, S. H. Tawfik, R. H. Baughman and A. J. Hart, *Science*, 2013, **339**, 535–539.
- 9 N. Karousis, N. Tagmatarchis and D. Tasis, *Chem. Rev.*, 2010, **110**, 5366–5397.
- 10 S. F. Liu, A. R. Petty, G. T. Sazama and T. M. Swager, *Angew. Chem., Int. Ed.*, 2015, **54**, 6554–6557.
- 11 S. F. Liu, L. C. H. Moh and T. M. Swager, *Chem. Mater.*, 2015, **27**, 3560–3563.
- 12 A. Maurin and M. Robert, *J. Am. Chem. Soc.*, 2016, **138**, 2492–2495.
- 13 P.-J. Wei, G.-Q. Yu, Y. Naruta and J.-G. Liu, *Angew. Chem., Int. Ed.*, 2014, **53**, 6659–6663.
- 14 I. Hijazi, T. Bourgeteau, R. Cornut, A. Morozan, A. Filoramo, J. Leroy, V. Derycke, B. Jousselmé and S. Campidelli, *J. Am. Chem. Soc.*, 2014, **136**, 6348–6354.
- 15 S. Jayakumar, H. Li, J. Chen and Q. Yang, *ACS Appl. Mater. Interfaces*, 2018, **10**, 2546–2555.
- 16 Q. Zhong, V. V. Diev, S. T. Roberts, P. D. Antunez, R. L. Brutchey, S. E. Bradforth and M. E. Thompson, *ACS Nano*, 2013, **7**, 3466–3475.
- 17 H. Imahori, T. Umeyama, K. Kurotobi and Y. Takano, *Chem. Commun.*, 2012, **48**, 4032–4045.
- 18 L. M. Arellano, M. Barrejon, H. B. Gobeze, M. J. Gomez-Escalonilla, J. L. G. Fierro, F. D'Souza and F. Langa, *Nanoscale*, 2017, **9**, 7551–7558.



- 19 G. Delport, F. Vialla, C. Roquelet, S. Campidelli, C. Voisin and J.-S. Lauret, *Nano Lett.*, 2017, **17**, 6778–6782.
- 20 T. Palacin, H. L. Khanh, B. Jousselme, P. Jegou, A. Filoramo, C. Ehli, D. M. Guldi and S. Campidelli, *J. Am. Chem. Soc.*, 2009, **131**, 15394–15402.
- 21 D. M. Guldi, H. Taieb, G. M. A. Rahman, N. Tagmatarchis and M. Prato, *Adv. Mater.*, 2005, **17**, 871–875.
- 22 D. M. Guldi, G. M. A. Rahman, M. Prato, N. Jux, S. Qin and W. Ford, *Angew. Chem., Int. Ed.*, 2005, **44**, 2015–2018.
- 23 G. Bottari, O. Trukhina, M. Ince and T. Torres, *Coord. Chem. Rev.*, 2012, **256**, 2453–2477.
- 24 F. D'Souza and O. Ito, *Chem. Commun.*, 2009, 4913–4928, DOI: 10.1039/b905753f.
- 25 S. Fukuzumi and T. Kojima, *J. Mater. Chem.*, 2008, **18**, 1427–1439.
- 26 L. Rodríguez-Pérez, S. Vela, C. Atienza and N. Martín, *Org. Lett.*, 2017, **19**, 4810–4813.
- 27 J. Lopez-Andarias, S. H. Mejias, T. Sakurai, W. Matsuda, S. Seki, F. Feixas, S. Osuna, C. Atienza, N. Martín and A. L. Cortajarena, *Adv. Funct. Mater.*, 2017, 1704031.
- 28 C. Ehli, G. M. A. Rahman, N. Jux, D. Balbinot, D. M. Guldi, F. Paolucci, M. Marcaccio, D. Paolucci, M. Melle-Franco, F. Zerbetto, S. Campidelli and M. Prato, *J. Am. Chem. Soc.*, 2006, **128**, 11222–11231.
- 29 S. Chichak Kelly, A. Star, M. V. P. Altoé and J. F. Stoddart, *Small*, 2005, **1**, 452–461.
- 30 G. Liu, F. Wang, S. Chaunchaiyakul, Y. Saito, A. K. Bauri, T. Kimura, Y. Kuwahara and N. Komatsu, *J. Am. Chem. Soc.*, 2013, **135**, 4805–4814.
- 31 A. F. M. M. Rahman, F. Wang, K. Matsuda, T. Kimura and N. Komatsu, *Chem. Sci.*, 2011, **2**, 862–867.
- 32 X. Peng, N. Komatsu, T. Kimura and A. Osuka, *ACS Nano*, 2008, **2**, 2045–2050.
- 33 X. Peng, N. Komatsu, T. Kimura and A. Osuka, *J. Am. Chem. Soc.*, 2007, **129**, 15947–15953.
- 34 X. Peng, N. Komatsu, S. Bhattacharya, T. Shimawaki, S. Aonuma, T. Kimura and A. Osuka, *Nat. Nanotechnol.*, 2007, **2**, 361–365.
- 35 R. S. Forgan, J.-P. Sauvage and J. F. Stoddart, *Chem. Rev.*, 2011, **111**, 5434–5464.
- 36 J. F. Stoddart, *Chem. Soc. Rev.*, 2009, **38**, 1802–1820.
- 37 J. F. Stoddart, *Angew. Chem., Int. Ed.*, 2017, **56**, 11094–11125.
- 38 J.-P. Sauvage, *Angew. Chem., Int. Ed.*, 2017, **56**, 11080–11093.
- 39 D. A. Leigh, *Angew. Chem., Int. Ed.*, 2016, **55**, 14506–14508.
- 40 E. M. Pérez, *Chem.-Eur. J.*, 2017, **23**, 12681–12689.
- 41 A. Mann Florian, J. Horlebein, F. Meyer Nils, D. Meyer, F. Thomas and S. Kruss, *Chem.-Eur. J.*, 2018, DOI: 10.1002/chem.201800993.
- 42 K. Miki, K. Saiki, T. Umeyama, J. Baek, T. Noda, H. Imahori, Y. Sato, K. Suenaga and K. Ohe, *Small*, 2018, **14**, 1800720.
- 43 S. Leret, Y. Pouillon, S. Casado, C. Navio, A. Rubio and E. M. Pérez, *Chem. Sci.*, 2017, **8**, 1927–1935.
- 44 A. Lopez-Moreno and E. M. Pérez, *Chem. Commun.*, 2015, **51**, 5421–5424.
- 45 A. de Juan, Y. Pouillon, L. Ruiz-Gonzalez, A. Torres-Pardo, S. Casado, N. Martín, A. Rubio and E. M. Pérez, *Angew. Chem., Int. Ed.*, 2014, **53**, 5394–5400.
- 46 A. López-Moreno, B. Nieto-Ortega, M. Moffa, A. de Juan, M. M. Bernal, J. P. Fernández-Blázquez, J. J. Vilatela, D. Pisignano and E. M. Pérez, *ACS Nano*, 2016, **10**, 8012–8018.
- 47 E. Martínez-Perinan, A. de Juan, Y. Pouillon, C. Schierl, V. Strauss, N. Martín, A. Rubio, D. M. Guldi, E. Lorenzo and E. M. Pérez, *Nanoscale*, 2016, **8**, 9254–9264.
- 48 R. Sure and S. Grimme, *J. Comput. Chem.*, 2013, **34**, 1672–1685.
- 49 A plot of the degree of functionalization vs. [U-por]₀ shows that the degree of functionalization does not have a linear relationship with the concentration of linear precursor, but instead shows a square hyperbolic shape, reminiscent of a 1 : 1 binding isotherm, reaching a maximum at about 30% (see Fig. S1†). Control experiments in which we mixed U-por with (6,5)-SWCNTs without Grubbs' catalyst, under otherwise identical conditions to the MINT reaction, yielded much lower functionalization (7%).
- 50 A. de Juan, M. Mar Bernal and E. M. Pérez, *ChemPlusChem*, 2015, **80**, 1153–1157.
- 51 These data taken together further prove that the supramolecular functionalization of the nanotubes by U-por, mac-por and/or linear oligomers formed *in situ* is minority.
- 52 S. T. Skowron, T. W. Chamberlain, J. Biskupek, U. Kaiser, E. Besley and A. N. Khlobystov, *Acc. Chem. Res.*, 2017, **50**, 1797–1807.
- 53 In DMF rather than D₂O/SDBS a blue-shift of the S₁₁ transitions for MINT-por relative to SWCNTs indicates a better debundling of SWCNTs.
- 54 Notably, we lack unambiguous proofs for the formation of the one-electron oxidized porphyrin, but we presume, as a sum of all spectroscopic evidence, charge transfer in small diameter SWCNTs. Here, the low intensity porphyrin transients, on one hand, and the strong SWCNT transients, on the other hand, are largely responsible for the lack of detection.
- 55 V. Sgobba, G. M. A. Rahman, D. M. Guldi, N. Jux, S. Campidelli and M. Prato, *Adv. Mater.*, 2006, **18**, 2264–2269.
- 56 D. M. Guldi and V. Sgobba, *Chem. Commun.*, 2011, **47**, 606–610.

