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Activating the surface and bulk of hematite photoanodes to improve solar water splitting†

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A simple electrochemical activation treatment is proposed to improve effectively the photoelectrochemical performance of Nb,Sn co-doped hematite nanorods. The activation process involves an initial thrice cathodic scanning (reduction) and a subsequent thrice anodic scanning (oxidation), which modifies both the surface and bulk properties of the Nb,Sn:Fe₂O₃ photoanode. First, it selectively removes the surface components to different extents endowing the hematite surface with fewer defects and richer Nb–O and Sn–O bonds and thus passivates the surface trap states. The surface passivation effect also enhances the photoelectrochemical stability of the photoanode. Finally, more Fe²⁺ ions or oxygen vacancies are generated in the bulk of hematite to enhance its conductivity. As a result, the photocurrent density is increased by 62.3% from 1.88 to 3.05 mA cm⁻² at 1.23 V_{RHE}, the photocurrent onset potential shifts cathodically by ~70 mV, and photoelectrochemical stability improves remarkably relative to the pristine photoanode under simulated sunlight (100 mW cm⁻²).

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1. Introduction

Photoelectrochemical (PEC) water splitting is a promising technology to convert solar energy directly into a storable hydrogen fuel using semiconductor photoelectrodes that can absorb sunlight and generate electron–hole pairs.¹ For practical applications, the photoelectrodes are required to have a high solar-to-hydrogen conversion efficiency (η_{STH}) greater than 10%, stability in an aqueous solution without photocorrosion, and low cost. Hematite is considered an ideal candidate photoanode material for PEC water splitting because of its natural abundance, good stability, an ideal bandgap of 2.1 eV for ample visible light absorption, and a suitable band position to drive water oxidation. However, it has some critical drawbacks that should be overcome to obtain a high η_{STH} , including a short hole diffusion length, poor conductivity, and a large overpotential for water oxidation.^{2,3}

A multitude of strategies have been developed to improve the PEC performance of hematite by addressing these shortcomings. Making a nanostructure (especially one-dimensional, small-diameter nanorods) can reduce the electron–hole recombination by shortening the actual diffusion distance of charge carriers. Elemental doping and/or creating oxygen

vacancies (V_{O}) are effective to improve its conductivity by enhancing the charge carrier density.^{4,5} Surface engineering can effectively overcome its large overpotential and low photocurrent density problems by removing the surface states through introducing a passivation layer and loading a co-catalyst.^{6–8}

In addition to these traditional modification strategies, some post-synthetic treatments can also significantly improve the performance of photoanodes for PEC water splitting. For example, Zou *et al.* reported a surface corrosion method in acid solution to reduce the overpotential of Ti⁴⁺-doped hematite by suppressing the back reaction.⁹ Li *et al.* reported an acid treatment method to improve the photocurrent density by suppressing electron–hole recombination in Sn⁴⁺-doped hematite.¹⁰ Jang *et al.* reported a re-growth strategy to reduce surface disorders of NiFeO_x-loaded hematite to achieve an onset voltage (V_{on}) of 0.45 V (*vs.* the reversible hydrogen electrode, RHE).⁶ It was also reported that the PEC performance of BiVO₄ was improved by ultraviolet irradiation¹¹ and electrochemical treatments.¹² Herein, we demonstrate a simple electrochemical process involving repeated reduction/oxidation to enhance the PEC performance of a Nb,Sn co-doped hematite nanorod (Nb,Sn:Fe₂O₃ NR) photoanode.

2. Results and discussion

2.1 Synthesis and characterization of Nb,Sn co-doped hematite nanorods

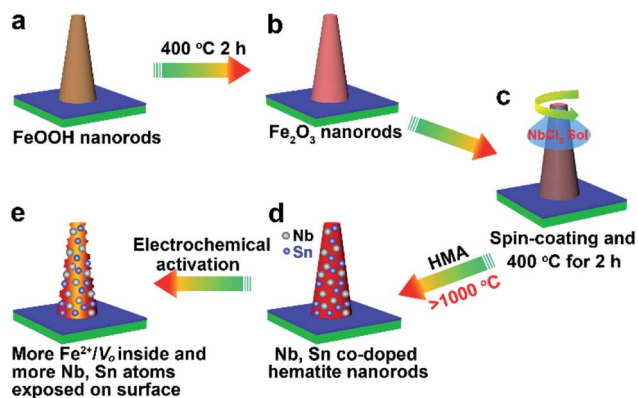
Scheme 1 shows schematics of the synthesis and electrochemical activation process of small Nb,Sn:Fe₂O₃ NRs synthesized on F:SnO₂ (FTO) glass substrates according to our

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Scheme 1 Synthesis and electrochemical activation process of a Nb,Sn:Fe₂O₃ NR photoanode.

previously reported method.^{13,14} In brief, ultrathin FeOOH nanorods are first grown on FTO by a hydrothermal method and turned into amorphous hematite nanorods at 400 °C for 2 h. Then, a low concentration of NbCl₅ solution is spin-coated and the second heating follows under the same conditions. Subsequently, hybrid microwave annealing (HMA) treatment is adopted to take advantage of its unique characteristics: (i) reaching extremely high temperature in a very short time of

a few minutes, (ii) preserving original nanoscale morphology under harsh fabrication conditions, and (iii) negligible damage of FTO substrates at a high temperature. These features of HMA are highly beneficial for the fabrication of efficient photoelectrodes, which the conventional thermal annealing cannot provide. Finally, the synthesized photoelectrode is electrochemically activated through an initial thrice cathodic scanning (reduction) and a subsequent thrice anodic scanning (oxidation). As discussed below, the activation treatment modifies both the bulk and surface properties of Nb,Sn:Fe₂O₃ NRs, leading to markedly enhanced PEC performance.

Our previous reports demonstrated that all the spin-coated Nb/Ta atoms were incorporated into the hematite lattice by the highly effective HMA.^{13,14} Simultaneously, Sn atoms would also diffuse into the hematite lattice from the bottom FTO substrate to complete the fabrication of Nb,Sn co-doped Fe₂O₃ nanorods. Electron energy loss spectroscopy (EELS) mapping (Fig. 1a–e) gives the spatial distribution of elements in hematite NRs, indicating that Fe, O and Nb are uniformly distributed in the whole nanorod and Sn has a little gradient from bottom (FTO source) to top. Furthermore, a high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) image (Fig. 1f) shows many bright atoms (circled), which must be the external doping atoms of Nb and Sn. Note that there are no attached nanoparticles on the nanorod surface, which

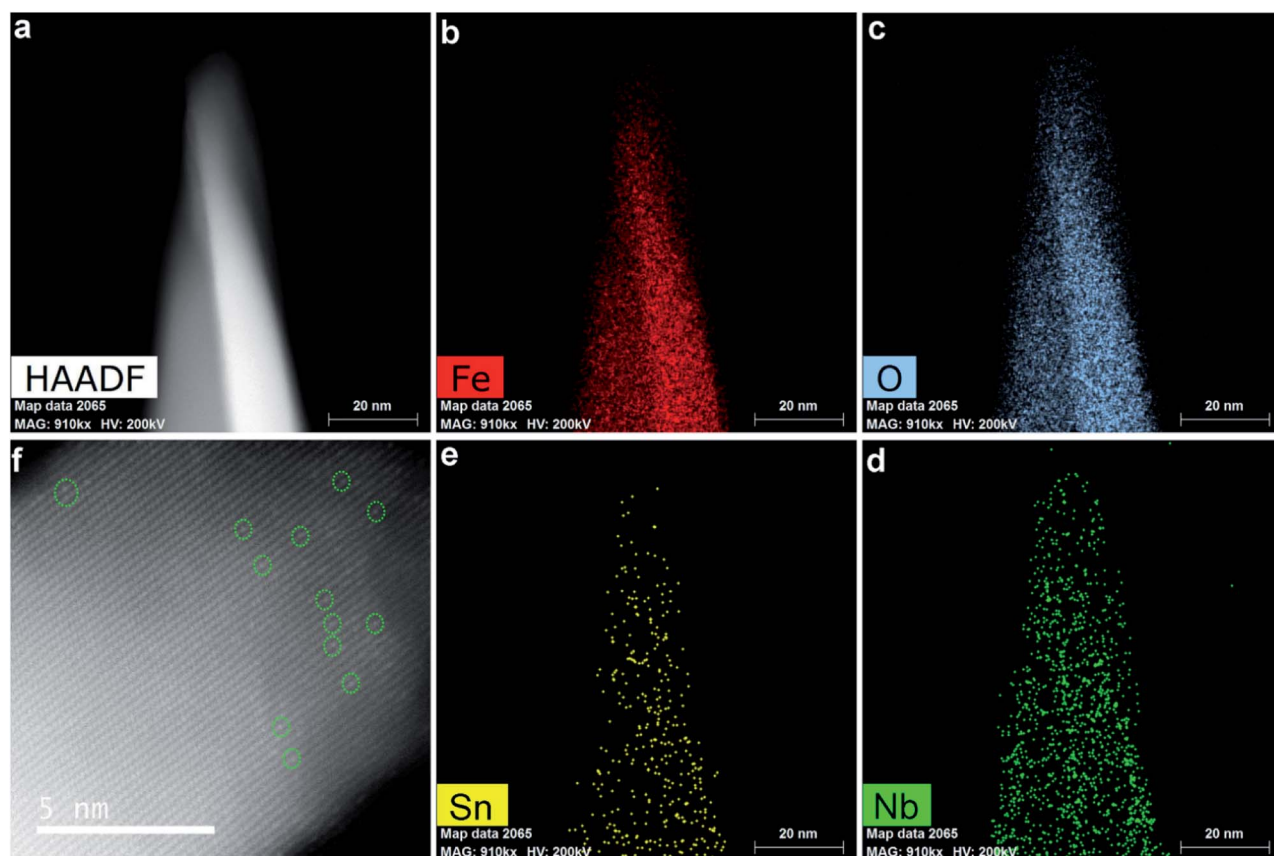


Fig. 1 STEM (a), corresponding EDX mapping ((b–e), Fe, O, Nb and Sn, respectively) and high-resolution HAADF-STEM (f) images of the synthesized Nb,Sn co-doped hematite nanorods.



demonstrates that all the spin-coated Nb atoms have been diffused into hematite lattices. Besides, the corresponding energy dispersive X-ray spectrum (EDX) shows that the concentrations of Nb and Sn in hematite NRs are 0.9 at% and 0.4 at%, respectively (Fig. S1, ESI†). The inductively coupled plasma optical emission spectrometry (ICP-OES) results demonstrate 5.5 wt% Nb (Fig. S2†), which agrees well with 0.9 at% Nb (~5.2 wt%) by TEM-EDX. The electrochemical activation process proposed here involves a thrice cathodic linear scanning followed by another thrice anodic linear scanning (detailed information in Fig. S3†). The treatment was conducted in 1 M KOH solution in the dark using a three-electrode cell with the Nb,Sn:Fe₂O₃ NR photoanode, Ag/AgCl (3 M NaCl), and Pt mesh as working, reference, and counter electrodes, respectively. The simple electrochemical process involving repeated reduction and oxidation enhances the PEC performance of the photoanode by modifying both the surface and bulk properties of hematite as detailed below.

Scanning electron microscope (SEM) images show no change in morphological features before (Fig. 2a) and after (Fig. 2b) the electrochemical treatment, both showing small nanorods with diameters of 20–50 nm. The activated sample also has a smooth surface similar to the pristine sample, and comparable surface areas measured using the adsorbed methylene blue dye (Fig. S4†). The X-ray diffraction (XRD) pattern after the activation treatment does not show any decreased peak intensities or shifted peak positions (Fig. 2c). Besides, Nb,Sn:Fe₂O₃ shows a little smaller diameter (increased full width at half maximum) and positive peak shifts relative to only Sn-doped hematite (Fig. S5†), indicating that Nb atoms from the surface coating are effectively doped into the hematite lattice while preserving the

original nanostructure. Moreover, absorption spectra before and after the treatment are nearly superimposable (Fig. 2d), thus ruling out any change in light harvesting capability by the activation process. High resolution transmission electron microscopy (HRTEM) images show that there are many defects (including dangling bonds, amorphous state Fe₂O₃, interstitial and vacancy defects) found on the surface of the hematite nanorod before activation (highlighted in Fig. 2e). In contrast, the surface is much cleaner after the activation (Fig. 2f), indicating that a type of etching takes place during the process. Both samples show well-defined HRTEM lattice fringes denoting preferential orientation of [012], indicating that the electrochemical activation does not change the high crystallinity induced by HMA, which is consistent with the XRD results.

2.2 Effects of the activation treatment on the surface and bulk properties of hematite

Surface-sensitive X-ray photoelectron spectroscopy (XPS) was combined with Ar⁺ etching to investigate the depth-dependent compositions of Nb,Sn:Fe₂O₃ photoanodes before and after the activation treatment. It should be noted that 300 s operation in our depth profiling achieves ~5 nm surface etching, which was estimated with a planar hematite film (synthesized by HMA) on silicon substrates. When the etching time exceeds 300 s, the intensity of the Nb 3d spectrum of the as-prepared Nb,Sn:Fe₂O₃ nanorod sample begins to decrease. As shown in Fig. 3a₀–d₀, there is almost no difference in XPS spectra of the pristine Nb,Sn:Fe₂O₃ NRs before and after surface etching for 300 s (except that Fe 2p and O 1s show a very limited decrease of intensity), demonstrating that the as-synthesized Nb,Sn:Fe₂O₃ NRs by HMA are homogeneous in composition from the surface

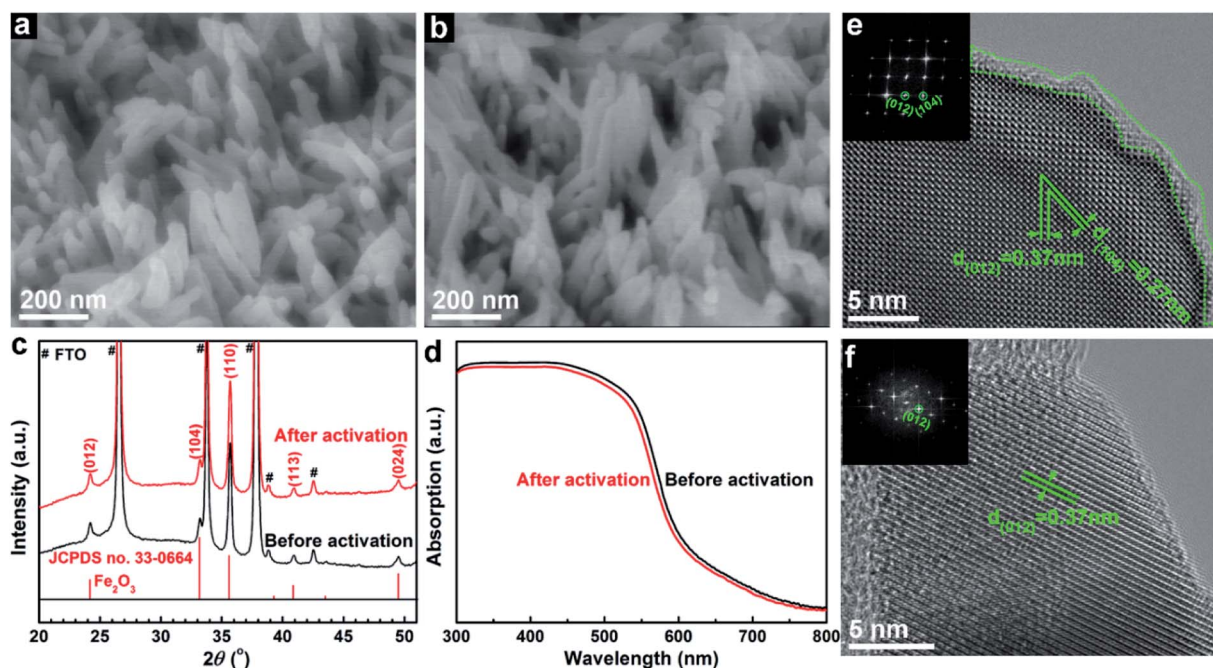


Fig. 2 SEM images ((a), before; (b), after), XRD (c) and UV-vis spectra (d) HRTEM (e and f) of the hematite photoanode before and after the electrochemical activation treatment. Insets (e and f) show the corresponding fast Fourier transform diffraction patterns (FFTs).



(before etching) to bulk (after etching) of the hematite. In contrast, the activated photoanode in Fig. 3a₁–d₁ shows significant differences. Upon etching, the intensity of Fe 2p and O 1s spectra decreases by 41.6% and 28.3%, while that of Nb 3d and Sn 3d spectra increases by 2.8% and 1.6%, respectively, which demonstrates that the activation process selectively removes the surface components of Nb,Sn:Fe₂O₃ to a different extent. As shown in Fig. S6,[†] the varied intensity was calculated from the integrated XPS peak areas. Thus, the activation treatment selectively removes Fe from the surface, which reflects its weakest bond energy: $D_{298}^{\circ}/\text{kJ mol}^{-1} = \text{Fe-O} (407) < \text{Sn-O} (528) < \text{Nb-O} (726)$.¹⁵ As a result, more Nb and Sn atoms remain on the surface than in the bulk after the treatment.

The electrochemical activation treatment brings another significant change in Fe 2p spectra (Fig. 3a₁). Both pristine and activated Nb,Sn:Fe₂O₃ samples show Fe 2p_{3/2} (~711.2 eV) and Fe 2p_{1/2} (~724.8 eV) peaks as typical for Fe³⁺ in Fe₂O₃.¹⁶ However, the surface of the activated sample shows a satellite peak of Fe³⁺ at 719.3 eV, which disappears after ~5 nm etching indicating the presence of more Fe²⁺ ions or V_O in the bulk of activated Nb,Sn:Fe₂O₃ compared with the pristine sample (Fig. 3a₀). In O 1s spectra (Fig. 3b₁), the surface of the activated sample exhibits a notable shoulder peak (dot circled) in the range of 531.9–533.7

eV, which can be attributed to defect sites, *e.g.* dangling oxo or hydroxyl groups usually observed around ~532.0 eV (ref. 17) and chemisorbed oxygen at 532.4–533.7 eV.¹⁸ This peak is not observed in the bulk after ~5 nm etching. Fig. 3c₁ displays typical peaks of Nb 3d_{5/2} at ~206.2 eV and Nb 3d_{3/2} at ~209.0 eV due to Nb⁵⁺, both on the surface and in the bulk. Besides, the binding energy of Nb 3d_{5/2} is between Nb₂O₅ (207.1 eV) and metallic Nb (202.4 eV), suggesting that Nb⁵⁺ ions are indeed doped into the Fe₂O₃ lattice.¹⁹ The Sn 3d spectra in Fig. 3d₁ exhibit two peaks centred at 486.4 eV for Sn 3d_{5/2} and 494.8 eV for Sn 3d_{3/2}. The binding energy of Sn 3d_{5/2} is between the typical values for SnO₂ (486.6 eV) and metallic Sn (484.6 eV), indicating that the Sn⁴⁺ dopants are also incorporated into the hematite lattice.²⁰ The change in the surface and bulk elemental structure before and after the electrochemical activation is schematically illustrated in Fig. S7.[†] Considering that the profiling depth of the pristine sample may not be deep enough due to the reduced thickness of the surface amorphous layer by activation, we extended the depth profiling time further to 400 and 500 s for the pristine sample (Fig. S8[†]). All the elements maintained exactly the same signals but with decreased intensities, which further demonstrates the homogeneous composition of the pristine sample from outside to inside.

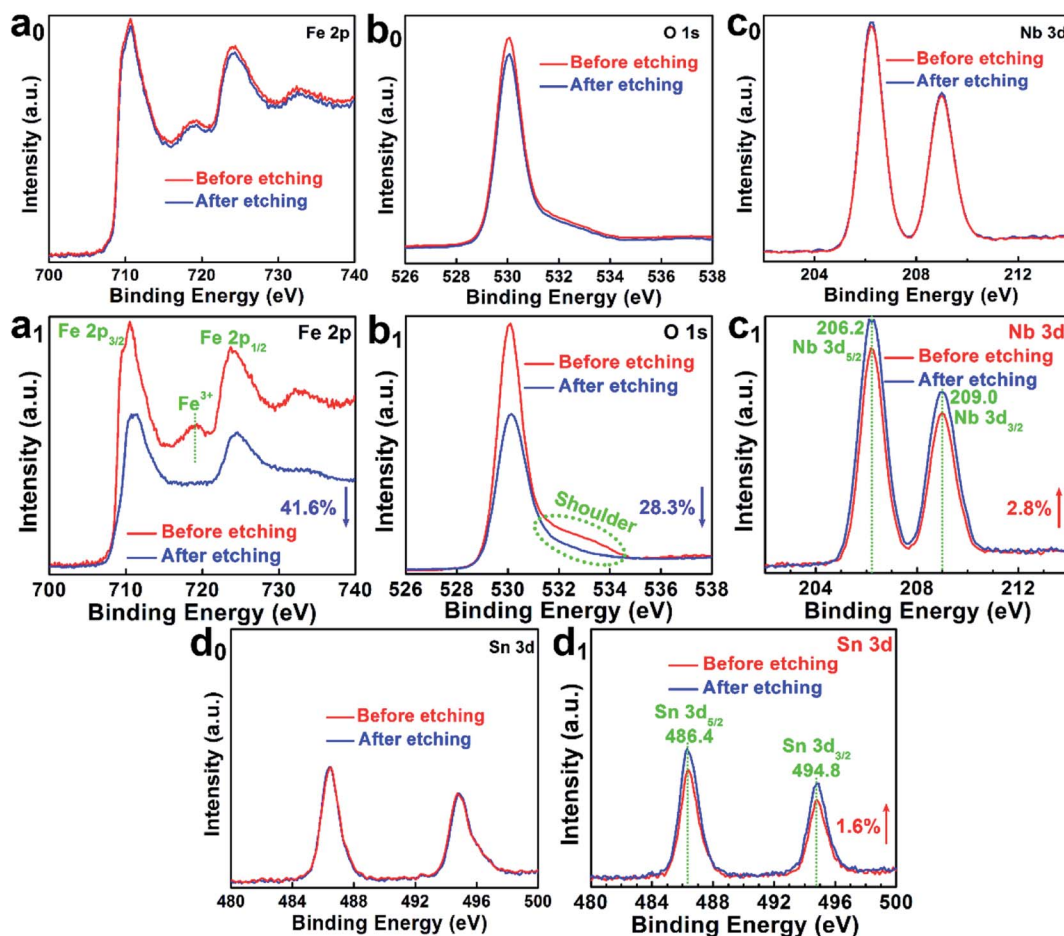


Fig. 3 XPS spectra of the pristine and activated Nb,Sn:Fe₂O₃ photoanode before and after surface etching (~5 nm), respectively. (a₀ and a₁) Fe 2p. (b₀ and b₁) O 1s. (c₀ and c₁) Nb 3d. (d₀ and d₁) Sn 3d.



2.3 Effects on photoelectrochemical water oxidation performance

After the electrochemical activation, the photocurrent density of PEC water oxidation reaches 3.05 mA cm^{-2} at $1.23 \text{ V}_{\text{RHE}}$ in 1 M KOH electrolyte under 1 sun (100 mW cm^{-2}) irradiation, which is significantly higher (by 62.3%) than that of the pristine photoanode (1.88 mA cm^{-2}) as shown in Fig. 4a. More significantly, the photocurrent onset potential (V_{on}) exhibited a modest cathodic shift of $\sim 70 \text{ mV}$, which was also similar to the shift ($\sim 50 \text{ mV}$) estimated by the first order derivative of the photocurrent density against the voltage²¹ (Fig. S9†). The modest shift of V_{on} should be related to the extremely thin surface passivation layer (only rich Nb–O and Sn–O bonds), which might not be a continuous layer and cannot accumulate many photoexcited holes on the surface. The measured photo-voltage (Fig. S10†) also shows $\sim 40 \text{ mV}$ increase from 160 mV (pristine) to 200 mV (after activation), which are consistent with previously reported values.²² It should be emphasized that the results were highly reproducible, which was consistently observed for the other repeated samples (Table S1†). By comparison, the pristine sample with a lower photocurrent exhibited a tendency to show a little higher extent of improvement after the activation. In fact, the activation process causes a little decreased surface area by 2.6% (Fig. S4†) by removal of some defective/amorphous part of hematite as indicated by HRTEM images in Fig. 1 and the XPS results in Fig. 3. In addition, the corresponding dark current of the water oxidation reaction is suppressed after the activation (inset of Fig. 4a). This behaviour implies a lower electrocatalytic activity, which was also observed in previous reports on acid-treated, doped-hematite and UV light-treated BiVO_4 .^{9–11} Obviously, the reduced surface area and the lower catalytic activity would not have contributed to the improved PEC performance in any positive manner. Then how does the activation

treatment enhance the PEC performance of the Nb,Sn:Fe₂O₃ photoanode?

As discussed, XPS combined with etching demonstrates that the activated Nb,Sn:Fe₂O₃ shows more Fe²⁺ ions or V_{O} in the bulk (etched surface) relative to the pristine photoanode (Fig. 3). As is well known, doping of high-valent Nb⁵⁺ and Sn⁴⁺ into the Fe₂O₃ lattice would enhance its n-type character by converting some Fe³⁺ into Fe²⁺ for charge compensation.²³ Electrochemical reduction during cathodic scanning of the activation process produces a sub-stoichiometric metal oxide by introduction of V_{O} ,²⁴ which is actually +2 charged and thus contributes to the n-type conductivity in the same manner as the metal doping.²⁵ It is difficult to maintain them near the surface because of contact with the electrolyte and water oxidation reaction during the anodic scanning, but they are well preserved in the bulk. In particular the highly doped Nb,Sn:Fe₂O₃ NRs are favorable to preserve Fe²⁺ ions or V_{O} in the bulk because their strong n-type character would make more free electrons available in the hematite lattice to suppress re-oxidation.

In order to demonstrate validity of the proposed sequential redox activation process, we tried the reverse sequence of thrice anodic scanning first followed by thrice cathodic scanning. The corresponding XPS spectra show decreased intensity of Fe 2p by only 25.2% (Fig. S11a†), which is $\sim 60\%$ that of Fe 2p spectra in Fig. 3a₁, indicating that the selective removal of Fe³⁺ ions from the surface is much less significant.

In addition, the satellite peak of Fe³⁺ at 719.3 eV still has a significant intensity remaining in the etched sample, indicating formation of fewer Fe²⁺ ions or V_{O} in the bulk. As a result, the corresponding PEC performance of the Nb,Sn:Fe₂O₃ photoanode activated by the reversed sequence exhibits only 39.8% improved photocurrent and 30 mV shift of V_{on} (Fig. S11b†). Therefore, cathodic scanning first and subsequent anodic scanning are essential for the efficient activation process, leading to significantly enhanced PEC performance, *i.e.* 62.3% higher photocurrent and $\sim 70 \text{ mV}$ shift of V_{on} as in Fig. 4a. Generally, the same element ions with different valences in equivalent points of a crystal lattice lead to a jumping mechanism for electronic conductivity²⁶ as in the present case of Nb,Sn:Fe₂O₃ NRs.

Nyquist plots of electrochemical impedance spectroscopy (EIS) in Fig. 4b for both activated and pristine electrodes are characterized by two semicircles at high (f_{H} , $100\text{--}2 \text{ kHz}$) and low (f_{L} , $2000\text{--}0.1 \text{ Hz}$) frequencies. They could be fitted by a simple equivalent model of two circuits (inset), which was first discussed by Hamann and Bisquert.^{27,28} The loop at f_{H} usually reflects the information of charge carriers in bulk hematite ($R_{\text{ct}}/C_{\text{bulk}}$), while the loop at f_{L} represents the information about the hematite/electrolyte interface ($R_{\text{trap}}/C_{\text{trap}}$). The fitting results are summarized in Table S2.† The sheet resistance of R_{s} (involving the electrolyte, external contact, and FTO resistances) is small and essentially constant.²⁹ The resistance R_{ct} is reduced by 33.4% by the activation treatment, probably by improved conductivity due to generation of more Fe²⁺ ions or V_{O} in the bulk of Nb,Sn:Fe₂O₃ NRs. More importantly, R_{trap} decreases significantly by 50.7% by selective removal of defects, unsaturated covalent bonds, and adsorbates on the surface while

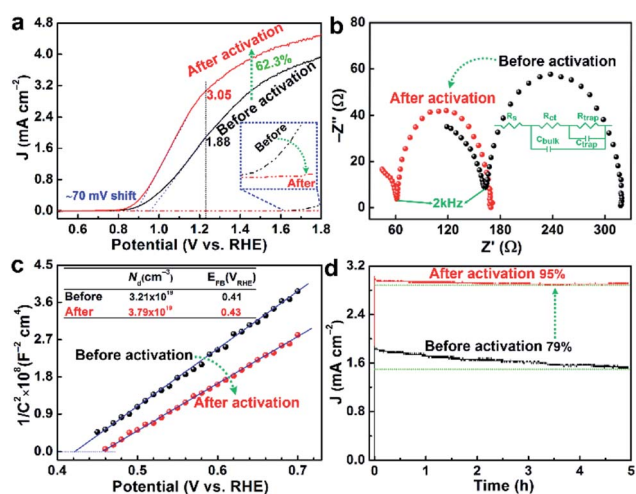


Fig. 4 Dark current (dashed) and photocurrent (solid) densities of PEC water oxidation in 1 M KOH electrolyte under 1 sun (100 mW cm^{-2}) irradiation (a), Nyquist plots of EIS (b), Mott–Schottky plots (c) and stability tests (d) of the Nb,Sn:Fe₂O₃ photoanode before and after the activation treatment.



maintaining substantial Nb–O and Sn–O bonds. Moreover, the capacitances of C_{bulk} and C_{trap} increase considerably after the activation.

Mott–Schottky plots were acquired from bulk capacitance values of EIS spectra, which gave the flat band potential (E_{FB} , the intercept values on the X -axis) and the donor density (N_{D} , inversely proportional to the slopes). In Fig. 4c, the electrochemical activation gives 18.1% increase of N_{D} and a positive shift of E_{FB} from 0.41 to 0.43 V_{RHE} , which are consistent with the reported values of hematite.³⁰ The increased N_{D} must be related to generation of more Fe^{2+} ions or V_{O} , in agreement with the above XPS/EIS results and a previous report for formation of Fe_3O_4 from Fe_2O_3 by electrochemical introduction of Fe^{2+} ions.³¹

On the surface of hematite, a range of different trap states including V_{O} , crystal defects, and hole accumulation sites ($\text{Fe}=\text{O}$) are proposed to be detrimental to PEC efficiency. As demonstrated above, there exist more Nb and Sn elements on the surface of Nb,Sn:Fe₂O₃ NRs after the activation, which would selectively passivate such states and consequently contribute to the enhanced photocurrent density and/or cathodically shifted V_{on} . Indeed, a very thin layer of Ta₂O₅ was demonstrated to have a good passivation effect on the hematite surface.³² Likewise, the mostly remaining Nb–O and Sn–O bonds on the hematite surface after activation should have a similar beneficial effect. In addition, they also contribute significantly to the stability of the photoanode because of their high bond energies. Indeed, Fig. 4d shows clearly enhanced stability after the activation treatment, maintaining 95% of initial photocurrent generation after 5 h of PEC water oxidation, whereas the pristine photoanode maintains only 79% of the initial activity. The results also confirm that the activated state remains stable and irreversible during the PEC water oxidation.

Steady-state photoluminescence spectroscopy (PL) and time-resolved PL were used to monitor the charge transfer dynamics in the activated Nb,Sn:Fe₂O₃ photoanode. Generally, bulk Fe₂O₃ exhibits little PL because of serious nonradiative recombination. However, its PL is highly dependent on surface quality, especially on surface trap density.³³ Therefore, PL would be enhanced when surface states are even partially eliminated by the activation process. The PL spectra of both activated and pristine Nb,Sn:Fe₂O₃ NRs in Fig. 5a show a weak emission-beginning peak at 575 nm (2.15 eV) and a relatively strong emission peak at 615 nm, which are consistent with previous reports of hematite.³⁴ In addition, the bare FTO treated under the same conditions as Nb,Sn:Fe₂O₃ synthesis shows no PL signal in the measured range (Fig. S12†). The enhanced PL of the activated sample is strong evidence that a part of surface trap states is passivated in the electrochemical activation process by selective removal of defects and retaining more Nb–O and Sn–O bonds.

In time-resolved PL (Fig. 5b), the activated sample shows a little higher tailing relative to that of the pristine sample, indicating that the activated sample has a longer charge carrier lifetime. The measured charge carrier lifetime denotes the time that takes for electron–hole pairs to decay to 1/e by radiative recombination.³⁵ Therefore, the longer lifetime means a slower

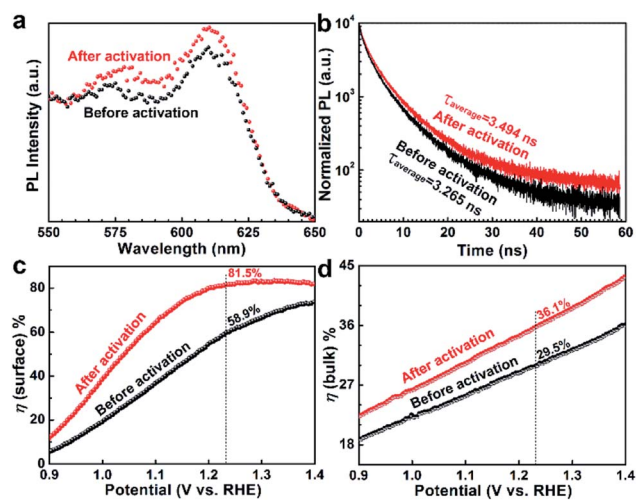


Fig. 5 Steady state PL spectra (a), time-resolved PL spectra (b), surface charge separation efficiency, η_{surface} (c), and bulk charge separation efficiency, η_{bulk} (d) of the Nb,Sn:Fe₂O₃ nanorod photoanode before and after the activation treatment.

recombination of the charge carriers. From the fitting of the PL decay kinetics, the amplitude weighted average lifetimes were $\tau_{\text{avg}} = 3.494$ ns for the activated sample and $\tau_{\text{avg}} = 3.265$ ns for the pristine sample (other parameters in Table S3†). In general, the short PL lifetime of hematite is attributed to the fast recombination between holes and electrons, which results from its short hole diffusion length. However, the activated sample shows 7% longer lifetime than that of the pristine sample, which must be related to the surface passivation effect by selective removal of surface trap sites and more Nb–O and Sn–O bonds retained on the surface by the activation process.

The surface (η_{surface}) and bulk (η_{bulk}) charge separation efficiencies were determined by comparing photo-oxidation currents of water and a hole scavenger (0.5 M H₂O₂) in the same electrolyte (1 M KOH). The detailed procedures are shown in Fig. S13.† The η_{surface} represents the fraction of holes at the interface that is successfully injected to the electrolyte to oxidize water without recombination on the semiconductor surface, while η_{bulk} is the fraction of holes that reach the interface of the semiconductor/electrolyte without recombination in the bulk. As in Fig. 5c, η_{surface} improves from 58.9% to 81.5% at 1.23 V_{RHE} after the electrochemical activation, indicating that the surface properties of hematite nanorods have been significantly improved by removal of surface trap sites and surface passivation by more Nb–O and Sn–O bonds retained. The η_{bulk} also improves from 29.5% to 36.1% at 1.23 V_{RHE} upon the electrochemical activation (Fig. 5d), indicating that the generated Fe^{2+} ions or V_{O} effectively enhances the electrical conductivity in the bulk of hematite. These results further support the aforementioned discussion on the effect of the activation treatment.

2.4 Application to other photoanodes

In order to test the general applicability of our electrochemical activation process, it was applied to two related materials. The first was Nb,Sn:Fe₂O₃@FeNbO₄ core–shell heterojunction NRs



that we studied in our previous work.¹³ As shown in Fig. S14,† the activation of the core-shell heterojunction shows very limited (only 6.2%) increase of photocurrent. The other repeated samples show similar behaviour (Table S4†). Because of high stability of the FeNbO₄ shell and strong n-type character of the core, we cannot take advantage of the beneficial effects of selective removal of the surface components to passivate surface trap sites and introduction of V_O/Fe²⁺ in lattices to increase the conductivity. The second test material was only Sn-doped hematite, which showed marked effects of the activation treatment. As shown in Fig. S15,† the photocurrent is improved by 44.4%, (from 1.24 to 1.79 mA cm⁻² at 1.23 V_{RHE} under 1 sun condition), which is comparable to the present Nb,Sn:Fe₂O₃ photoanode (62.3% increase). The other activated samples (Table S5†) also show reproducible results and the same tendency with the Nb,Sn:Fe₂O₃ electrode (higher extent of improvement on the poorer pristine sample). This is reasonable because the poor pristine samples (prepared under the same conditions) usually have more surface states, and most of them would be passivated by the activation. The results confirm that our activation strategy is applicable to improve the activity of hematite-based photoanodes in general, but is not very effective for hematite with a stable protective layer, for which the selective removal of the surface components is difficult.

3. Conclusions

In summary, we have demonstrated a simple and effective electrochemical activation process that improves the PEC water oxidation performance of the Nb,Sn:Fe₂O₃ photoanode in terms of photocurrent density, onset potential, and stability. The activation process involves initial thrice cathodic scanning followed by thrice anodic scanning, which makes three main modifications of hematite. First, it selectively removes the surface components to different extents endowing the hematite surface with fewer defects and richer Nb–O and Sn–O bonds and thus passivates the surface trap states. The surface passivation effect also enhances the PEC stability of the photoanode. Finally, more Fe²⁺ ions or V_O is generated in the bulk of hematite to enhance its conductivity. These changes in the bulk and on the surface of hematite are responsible for the greatly enhanced PEC performance by improving charge separation efficiencies both on the surface (η_{surface}) and in the bulk (η_{bulk}). The electrochemical activation strategy proposed here could be extended to a number of other metal oxide semiconductors including WO₃, TiO₂, BiVO₄ and ZnO for significantly improved PEC water splitting activity.

4. Experimental section

4.1 Fabrication of ultrathin FeOOH nanorods on FTO substrates

Conductive FTO substrates (PECTM 8, 6–9 Ω , Pilkington) of 25 \times 50 mm² were cleaned by ultrasonication for 30 min in detergent, ethanol, and acetone, respectively. The FTO surface becomes sufficiently hydrophilic by this cleaning process. The FeOOH nanorods grow on FTO at 100 $^{\circ}\text{C}$ for 2 h using 100 mL of

an aqueous solution containing 4.0 g FeCl₃·6H₂O (Sigma Aldrich, $\geq 99\%$), 200 μL HNO₃ (Sigma Aldrich, 70%) and 8.4 g NaNO₃ (Sigma Aldrich, $\geq 99\%$). The obtained yellow-colored FeOOH nanorod film after cooling down was rinsed with abundant deionized water.

4.2 Hybrid microwave annealing (HMA)

The prepared samples (12.5 \times 12.5 mm²) were put on graphite powder (60 mL graphite as a susceptor) in a Pyrex beaker (100 mL) and treated in a microwave oven (2.45 GHz, 1000 W) for 2 min (100% power).

4.3 Physical characterization

The morphology of the samples was acquired using a field-emission scanning electron microscope (FESEM-S4800, HITACHI). High-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images and corresponding electron energy loss spectroscopy (EELS) mapping were taken using a FEI Titan3 G2 60-300 microscope equipped with a double-sided Cs corrector operating at 200 kV. High resolution TEM (HRTEM) images in bright field were obtained using a Cs-corrected high-resolution transmission electron microscope (JEOL, JEM-2100F, 200 kV). X-ray diffraction (XRD) spectra were acquired using a PW3040/60 X'per PRO, PANalytical, using Cu-K α ($\lambda = 1.54056 \text{ \AA}$) radiation, an accelerating voltage of 40 kV and a current of 30 mA. Ultraviolet-visible absorbance was obtained using a UV-vis spectrometer (UV-2401PC, Shimadzu). X-ray photoelectron spectroscopy (XPS) was carried out on a Thermo-Fisher machine (XPS, ESCALAB 250XI) using an Al K α source. Inductively coupled plasma optical emission spectrometry (ICP-OES) was obtained on a Varian machine (700-ES). Steady state photoluminescence (PL) measurements were conducted on a spectrofluorometer (FLS920) with an excitation wavelength of 450 nm using a 450 W xenon lamp excitation source and a photomultiplier tube detector. Time-resolved PL spectra were measured using a time-correlated single-photon counting (TCSPC) setup (FluoTime 300). The excitation source was a 450 nm continuous-wave and pulsed diode laser head (LDH-D-C-450) coupled with a laser diode driver (PDL 820) with a pulse width of <70 ps and a repetition rate of 196 kHz to 40 MHz. Each exponential decay curve was deconvoluted using the associated fitting software (FluoFit) to calculate the time constant associated with each curve.

4.4 (Photo)electrochemical measurements

All (photo)electrochemical measurements were conducted on a potentiostat (IviumStat, Ivium Technologies) in a three electrode system using a hematite photoanode, Ag/AgCl (3 M NaCl), and Pt mesh as working, reference, and counter electrodes, respectively, in 1 M KOH electrolyte under 1-sun condition (100 mW cm⁻²) using a solar simulator (91160, Oriel) with an AM 1.5 G filter. Note that the exposed surface area of all the prepared photoanodes was 1 cm² and the electrolyte was slowly stirred during the measurements. All the potentials vs. the Ag/AgCl reference electrode were converted to the potentials vs. the reversible hydrogen electrode (RHE) according to the Nernst



equation: $E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 0.059 \text{ pH} + E_{\text{Ag/AgCl}}^{\circ}$ ($E_{\text{Ag/AgCl}}^{\circ} = 0.1976$ at 25 °C). The potential was swept in the range of 0.5–1.8 V_{RHE} with a scanning rate of 20 mV s^{-1} . Electrochemical impedance spectroscopy (EIS) was carried out at 1.23 V_{RHE} under simulated 1-sun condition with a frequency range of 0.1 Hz to 100 kHz and the data were fitted by Z-view software. Mott–Schottky plots were obtained by sweeping a 0.1–1.0 V_{RHE} range with an AC frequency of 1000 Hz under dark conditions. The charge carrier density, N_{D} , can be extracted and is inversely proportional to the slope as shown in eqn (1):

$$\left(\frac{A_{\text{s}}}{C_{\text{bulk}}}\right)^2 = \frac{2}{q\epsilon_{\text{r}}\epsilon_0 N_{\text{D}}} \left(V - E_{\text{FB}} - \frac{k_{\text{B}}T}{q}\right) \quad (1)$$

with $\left(\frac{A_{\text{s}}}{C_{\text{bulk}}}\right)$ being the surface area-corrected space charge capacitance, V the applied potential, E_{FB} the flat band of the electrode, $k_{\text{B}} = 1.38 \times 10^{-23} \text{ J K}^{-1}$, $T = 298 \text{ K}$, $q = 1.602 \times 10^{-19} \text{ C}$, $\epsilon_0 = 8.85 \times 10^{-12} \text{ C}^2 \text{ J}^{-1} \text{ m}^{-1}$, and $\epsilon_{\text{r}} = 32$ for hematite.

Conflicts of interest

There are no conflicts to declare.

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