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1. Introduction

Recently, semiconductor photocatalysis as an ideal green technology has been extensively employed to deal with the energy shortage and environmental pollutions because of its direct utilization of solar energy and environmental friendliness.¹⁻⁴ Although the $TiO₂$ semiconductor has attracted worldwide attention owing to its low price, nontoxicity and good stability against photocorrosion, $5-7$ it can only capture ultraviolet (UV) light due to its relatively large bandgap (\sim 3.2 eV).^{8,9} UV light only comprises \sim 4% of the solar spectrum, while the visible light makes up the greater part of sunlight.¹⁰ Thus, more and more visible-light-driven photocatalysts have been designed and synthesized to maximize the sunlight utilization efficiency, such as α -Fe₂O₃,¹¹ Bi₂MoO₆,¹² Ag₃PO₄ (ref. 13) and SnS₂.^{14,15}

Among the newly developed photocatalysts, Bi-based oxides such as BiOBr, BiOI and BiOCl have been extensively investigated due to their high stability, low price and outstanding photocatalytic activity.¹⁶–¹⁹ Among the BiOX photocatalysts, BiOI, a p-type semiconductor, has the smallest band gap (1.6– 1.9 eV) and strongest visible-light response capacity.²⁰–²²

Fabrication of a novel BiOI/KTaO₃ p-n heterostructure with enhanced photocatalytic performance under visible-light irradiation

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In this study, a series of BiOI/KTaO₃ p-n heterojunctions were prepared via a facile in situ chemical bath strategy. The photocatalytic properties of the catalysts was tested by the degradation of Rhodamine B (RhB) and phenol under visible light irradiation. The BiOI/KTaO₃ composites exhibited improved photocatalytic efficiency compared to the individual catalysts. In particular, 54 wt% BiOI/KTaO₃ displayed the highest photocatalytic activity since it degraded 98.6% RhB within 30 minutes, while only 68.1% RhB was degraded over pure BiOI under identical conditions. In addition, the reaction kinetic constant of RhB degradation over 54 wt% BiOI/KTaO₃ was approximately 2.56 and 115-fold larger than those of pure BiOI and KTaO $_3$, respectively. The results of PL, photocurrent and EIS indicated that the improved photocatalytic efficiency could root in the p-n junction formed between BiOI and KTaO $_3$, which was conducive to the separation and migration of photo-generated carriers. Furthermore, a free-radical capture experiment illustrated that h⁺ and $^{\circ}O_2^{-}$ were the key factors in the photodegradation of RhB. PAPER

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Therefore, BiOI has been widely utilized in the photocatalytic removal of antibiotics and refractory pollutants $22-25$ and the photoreduction of $CO₂$.²⁶ Up to now, researchers have successfully fabricated many BiOI microstructures with various morphologies, such as nanosheets,¹⁸ nanoflowers^{25,27} and microspheres.20,28 Although extensive efforts have been made towards the fabrication of BiOI configurations, the photocatalytic properties of individual BiOI was still not satisfactory because of the short lifetime of photoexcited charge carriers.²¹ Hence, many strategies have been developed to make BiOI applicable in practical use.^{26,29} One of the most effective alternatives was to combine BiOI with another semiconductor to form heterojunctions. In particular, building p–n heterostructures can effectively promote the transfer of the photoexcited carriers owing to the inducement of inner electric field and closely contacted interfaces.^{30,31} For example, Wen et al. successfully prepared $p-n$ heterostructure BiOI/SnO₂ and found that the hybrids displayed an outstanding photocatalytic activity towards the degradation of refractory pollutants.¹⁸ Lately, Xia et al. reported a novel n-p heterostructure SrTiO₃/ BiOI composite and its excellent photocatalytic properties for crystal violet decomposition under simulated solar light irradiation.²⁹ Thus, constructing a p–n heterojunction by coupling BiOI to an n-type semiconductor with appropriate energy band configuration was a promising method to obtain efficient visible-light-activated photocatalysts.

Moreover, as an important class of photocatalysts, perovskite-structured tantalates have been widely employed in both the mineralization of organic pollutants and water

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splitting due to their photoactivity, chemical stability, and nontoxicity.³²–³⁴ Among the tantalate photocatalysts, potassium tantalate (KTaO₃), a typical n-type semiconductor, was shown to be a promising photocatalytic material.³⁵ However, the wide band gap $(\sim3.6 \text{ eV})$ and unsatisfying quantum efficiency hampered its practical applications.³⁶ Therefore, numerous strategies, such as doping of foreign elements,³⁷ construction of solid solutions,³⁸ and building heterojunctions,^{39,40} have been developed to strengthen the photocatalytic activity. For example, Yong et al. synthesized $g-C_3N_4/KTaO_3$ composite and found that it displayed a largely enhanced photocatalytic property for RhB degradation under visible-light irradiation.³⁶ However, $KTaO₃$ -based heterostructures with an outstanding photocatalytic activity and high stability are still rare. Building p–n heterojunction $BiOI/KTaO₃$ hybrid materials would be an efficient way to form an ideal composite photocatalyst with high visible-light activity. Unfortunately, as far as we know, the research focused on the $BiOI/KTaO₃$ photocatalyst has not been reported. RSC Advances Computer Article computer Article 2020. The published on 17 March 2022 Articles. The published on 17 March 2022 Articles Article is likely and the summarized of the summarized photosteric interactions, the me

In this study, for the first time, we fabricated p-n heterojunction BiOI/KTaO₃ composites through a facile strategy. The photocatalytic efficiency of the $BiOI/KTaO₃$ heterostructure was examined by RhB degradation with visible-light irradiation (λ > 400 nm). The outcomes showed that the BiOI/KTaO₃ composites were high-efficiency visible-light-activated photocatalysts to remove organic pollutants. The BiOI/KTaO₃ heterojunctions showed a much superior photocatalytic activity towards RhB removal compared to BiOI and $KTaO₃$ alone. Furthermore, the possible photocatalytic mechanism behind the improved photocatalytic efficiency has been proposed and discussed in detail.

2. Experimental

2.1 Preparation of $KTaO₃$

The KTaO₃ nanocubes were fabricated via a hydrothermal process. In brief, 1.105 g Ta₂O₅ was dissolved in a 25 ml KOH aqueous solution (15 mol ${\rm l}^{-1})$ under constant magnetic stirring for 1 hour. Subsequently, the mixed solution was reacted at 150 \degree C for 5 hours. Ultimately, the product was cleaned with deionized water and anhydrous ethanol, and then dried at 60 °C for 12 hours to get the $KTaO₃$ sample.

2.2 Synthesis of BiOI/KTaO₃

The BiOI/KTaO₃ heterostructures were fabricated through a facile in situ chemical bath method. First, 300 mg of KTaO₃ was dispersed into 40 ml of distilled water under ultrasonication for 0.5 hour and then stirred magnetically for 10 min to get a well-dispersed suspension, marked as A. Second, 20 ml of ethylene glycol containing 1 mmol $Bi(NO₃)₃·5H₂O$ was dropwise added to A; the mixed solution was marked as B. Third, 10 ml of deionized water containing 1 mmol of KI was added to B with stirring violently for 0.5 hours; it was marked as C. Subsequently, C was reacted in a water bath at 80 \degree C for 2 hours. Finally, the sediment was cleaned with deionized water and anhydrous ethanol, and then dried at 60 \degree C overnight to obtain

54 wt% BiOI/KTaO₃ samples. Other BiOI/KTaO₃ composites with different contents of BiOI (37 wt% and 70 wt%) were fabricated by varying the amounts of $Bi(NO₃)₃·5H₂O$ and KI. Pure BiOI was synthesized via a similar way without the addition of $KTaO₃$.

2.3 Characterization

The phases of all the samples were characterized using an X-ray diffractometer (XRD, D8 ADVANCE, Bruker, Germany) with Cu K_{α} radiation. The field-emission scanning electron microscopy (FESEM, SU8220, Hitachi, Japan) and transmission electron (TEM) and high-resolution TEM (HRTEM) (JEM-2100) were used to observe the microstructure and element mapping of the samples. A UV visible absorption spectrum (DRS) was recorded using a PerkinElmer Lambda 950 spectrometer using BaSO₄ powder as reference. The element state of $BiOI/KTaO₃$ was characterized via X-ray photoelectron spectroscopy (XPS) using an ESCALAB250Xi spectrometer. Photoluminescence (PL) spectra were recorded on a PLS920 fluorescence spectrometer at room temperature.

2.4 Photocatalytic experiment

The photocatalytic property of the as-prepared samples was measured through detecting the degradation of RhB (10 mg l^{-1}) and phenol (5 mg I^{-1}) under visible-light irradiation. A 500 W Xe lamp equipped with a 400 nm filter was used as the light source (180 mW cm^{-2}) . In detail, 0.05 g of samples was put into a 100 ml RhB or phenol solution. Then, the suspension was stirred in dark intensely for 1 hour to get absorption–desorption equilibrium. After that, the stirred suspensions were exposed to the irradiation, 4 ml liquid was taken out and centrifuged at a certain timespan. Then, a spectrophotometer (UV-6000PC) was used to detect the concentration changes of RhB or phenol. The characteristic peaks were at 553 nm and 270 nm for RhB and phenol, respectively.

2.5 Photoelectrochemical test

Electrochemical detection was carried out on a photoelectrochemical workstation (CHI 760E, Shanghai Chenhua, China) that was equipped with a standard three-electrode. Electrolyte was prepared with 0.2 M $Na₂SO₄$, and the working electrode was manufactured by the following steps: first, 0.03 g sample was added into 100 μ l of ethyl alcohol with 20 μ l of Nafion; second, the photocatalyst was ground to form a pulp; third, the slurry photocatalyst was deposited on a fluoride-tin oxide (FTO) glass. The Ag/AgCl electrode was used as the reference electrode, while a Pt wire as the counter electrode.

3. Results and discussion

3.1 Characterization

The phase structure of those as-formed catalysts was investigated via XRD. Fig. 1 illustrates that all of the diffraction peaks for BiOI can be well assigned to the pure tetragonal BiOI phase (JCPDS no: $10-0445$) and those of KTaO₃ perfectly assigned to the cubic phase of $KTaO₃$ (JCPDS no: 38-1470). In addition, the

Fig. 1 XRD patterns of BiOI, $KTaO_z$ and BiOI/KTaO_z composites.

peaks in the BiOI/KTaO₃ composites include both the diffraction peaks of BiOI and $KTaO₃$, revealing that the two phases existed in $BiOI/KTaO₃$ heterostructures. There was no other impurity peaks in the composites, thus indicating the high purity of the BiOI/KTaO₃ composites.

FESSEM was used for determining the microstructure of pure KTaO₃, BiOI and BiOI/KTaO₃ heterostructures. As shown in Fig. 2a, pure $KTaO₃$ presented irregular nanocubes with smooth surfaces, and the size of those cubes was about 300 nm. Moreover, pure BiOI (Fig. 2b) showed the structure of nanosheets. With respect to the BiOI/KTaO₃ composites (Fig. 2c), a large amount of BiOI nanosheets attached tightly to

the surface of $KTaO₃$ nanocubes, and Fig. 2d presents the enlarged SEM image of the BiOI/KTaO₃ composite. It is worth noting that the size of the BiOI nanosheets became smaller after being contacted with KTiO₃. The reason may be attributed to the fact that the growth of BiOI could be restricted by $KTaO₃$ nanocube. To further demonstrate the heterojunction established between $KTaO₃$ and BiOI, elemental mapping was introduced. The distribution of elements of 54 wt% BiOI/ $KTaO₃$ is shown in Fig. 3(a)–(f); the I, K, Ta, Bi and O maps were detected, which illustrated that the composite consisted of BiOI and KTaO₃.

The detailed interface contact of $BiOI/KTaO₃$ was detected by TEM and HRTEM. Fig. 4a displays that the BiOI nanosheets are fixed on the surface of the $KTaO₃$ nanocubes. Fig. 4b shows the HRTEM image of $BiOI/KTaO₃$, and there are two sets of lattice fringes in different directions. The interplanar crystal spacing of 0.2675 nm was indexed to the (111) crystallographic plane of BiOI, while 0.3898 nm belonged to plane (100) of KTaO₃, which was consistent with the previous studies.^{41,42}

To analyze the elemental valence state of the composites, XPS measurement was carried out. As shown in Fig. 5a, Bi, O, I, K and O coexisted in the BiOI/KTaO₃ composite. The peak of C 1s at 284.8 eV is due to the adventitious hydrocarbon produced by the XPS instrument.⁴³ Fig. 5b showed double peaks at the binding energies of 619.1 and 630.6 eV, which are indexed to I $3d_{5/2}$ and I $3d_{3/2}$,⁴⁴⁻⁴⁶ respectively. Two peaks for K 2p at 292.2 and 294.9 eV (Fig. 5c) belonged to K $2p_{3/2}$ and K $2p_{1/2}$, respectively. Two distinct peaks at 26.2 and 28.4 eV are assigned to Ta $4f_{7/2}$ and Ta $4f_{5/2}$,

Fig. 2 SEM images of KTaO₃ (a), BiOI (b) and BiOI/KTaO₃ heterostructures (c and d)

Fig. 3 Elemental mapping of 54 wt% BiOI/KTaO₃ composite (a-f).

respectively; it was obvious that Ta existed as Ta^{5+} in KTaO₃ (ref. 47) (Fig. 5d). Fig. 5e shows the peaks of Bi $4f_{5/2}$ and Bi $4f_{7/2}$ at 164.5 and 159.2 eV, $44,45,48$ respectively. The peak for O 1s at

530.2 eV came from BiOI or KTaO₃ as O^{2-} , while the other peak at 531.6 eV (Fig. 5f) belonged to the hydroxyl groups.^{44,48} The XPS results indicated the coexistence of BiOI and KTaO₃.

Fig. 4 TEM (a) and HETEM (b) images of BiOI/KTaO₃ composite.

Fig. 5 XPS patterns of 54 wt% BiOI/KTaO₃ composite: (a) survey; (b) I 3d; (c) K 2p; (d) Ta 4f; (e) Bi 4f; and (e) O 1s.

The optical absorption property of the BiOI, KTaO₃ and BiOI/ $KTaO₃$ composites was characterized via DRS. Optical absorption coefficient satisfied the following formula:

$$
\alpha h\nu = A(h\nu - E_{\rm g})^{n/2}
$$

where α , h , v , E_g and A refer to absorption coefficient, Planck's constant, light frequency, energy gap and constant,^{49,50} respectively. From Fig. 6a, the absorption edge of pure $KTaO₃$ was at about 350 nm, while that of BiOI was at \sim 680 nm. The absorption edge of the BiOI/KTaO₃ composite is located between 600 and

680 nm. Moreover, with the increase in the BiOI content, the absorption edge of the composite was extended. Compared to the pure KTa O_3 , the composite material exhibited a red-shift. In addition, n value was decided by the light transition mode of the semiconductor, $n = 1$ for the direct transition, while $n = 4$ for the indirect transition. The optical transitions model of $KTaO₃$ is direct, while that of BiOI is indirect. On the basis of the Tauc plot, the energy gaps of pure $KTaO₃$ and BiOI are about 3.75 eV (Fig. 6b) and 1.76 eV (Fig. 6c), respectively, which are similar to those in the previous literatures.^{18,51}

Fig. 6 (a) UV-vis absorption spectrum of samples, (b) bandgap energy of KTaO₃ and (c) BiOI.

3.2 Photocatalytic activity

The photocatalytic efficiency of the BiOI/KTaO₃ hybrid materials was examined by decomposing RhB in the presence of visible light. The adsorption property of the sample has a signicant effect on the degradation process; thus, the adsorption performance of as-obtained catalysts for RhB degradation was first studied and displayed in Fig. 7(a). Clearly, the adsorption efficiencies of the hybrid materials increased along with the increase in the BiOI content, which could be ascribed to the relatively high surface area of BiOI. In addition, the adsorption–desorption equilibrium was achieved in an hour; hence, the adsorption made almost no difference on the following photocatalytic experiments. The RhB photodegradation with the various samples is displayed in Fig. 7(b). Obviously, pure KTaO₃ has a poor photocatalytic capacity and only about 3% of the dye was degraded, while about 68.1% of RhB was degraded by BiOI within 30 min. Compared to pure catalysts, the BiOI/KTaO₃ composites showed superior photocatalytic efficiency, which can be ascribed to the effective interfacial charge transfer due to the formation of heterojunctions. Furthermore, it is suggested that the amount of BiOI in composites played a significant part in the photodegradation process. The degradation efficiency of RhB over BiOI/KTaO₃ heterojunctions increased first, and then decreased along with

the increase in the BiOI content. When the mass percentage of BiOI in the composite was 54 wt%, BiOI/KTaO₃ heterojunctions exhibited the optimal photocatalytic activity as it degraded \sim 98.6% of RhB in 30 min, which was 1.45-fold larger than that of single BiOI.

For the purpose of investigating the degradation kinetics of RhB, the pseudo-first-order model was introduced. Fig. $7(c)$ shows the fitted curve of the measured data. The result satisfied the pseudo-first-order model. The kinetic constant was determined by the following equation: $ln(C_0/C_t) = kt^{52}$ Herein, C_0 and C_t denote the concentrations of RhB when time is 0 and t min, respectively, and k is the constant of kinetics. Fig. $7(c)$ displays that the slope of the BiOI/KTaO₃ composite is much higher than that of pure samples. Moreover, the k values of all the samples are calculated and shown in Fig. $7(d)$. The k value of 54 wt% BiOI/KTaO₃ was nearly 0.115 min^{-1} , while that of BiOI, 37 wt% BiOI/KTaO₃, 70 wt% BiOI/KTaO₃ and KTaO₃ was about 0.045 min⁻¹, 0.059 min⁻¹, 0.074 min⁻¹ and 0.001 min⁻¹, respectively. In addition, the k value of 54 wt% BiOI/KTaO₃ was about 2.56 and 115 times as much as BiOI and KTaO₃.

The photocatalytic degradation of the non-dye compound was also studied with that of the as-fabricated samples. Fig. 8 displays the photodegradation of phenol under the irradiation of visible light. Clearly, the degradation efficiency of phenol by all catalysts followed the order: 54 wt% BiOI/KTaO₃ > 70 wt%

Fig. 7 (a) Absorption ability evaluation of the as-prepared samples for RhB in dark condition; (b) photocatalytic degradation efficiencies of RhB; (c) the pseudo-first-order reaction kinetics for RhB degradation; (d) the degradation rate constants.

BiOI/KTaO₃ > 37 wt% BiOI/KTaO₃ > BiOI > KTaO₃. The result was consistent with the outcomes of RhB decomposition under visible-light irradiation. The 54 wt% BiOI/KTaO₃ composites also exhibited the highest photocatalytic capacity for the removal of phenol among the as-obtained catalysts.

The stability of the catalyst is an important factor in practical applications. Herein, a cycle experiment was carried out using

Fig. 8 Time profiles of the photocatalytic degradation of phenol under visible light irradiation.

the 54 wt% $BiOI/KTaO₃$ composite. Fig. 9(a) displays that the photodegradation efficiency of dyes still reached 91.1% after three cycling runs. Besides, the photocatalyst was collected after the cycle experiment, and the phase structure of the catalyst before and after the cycle experiment is presented in Fig. 9(b). No other changes could be found from the XRD result, indicating that the structure of the composite was stable.

3.3 Mechanism of the improved photocatalytic efficiency

To explore the main active species in the photodegradation process, free-radical capture experiments were performed. Benzoquinone (BQ) was used as the scavenger for $^{\cdot}O_{2}^{-}$,⁵³ while triethanolamine (TEOA) and isopropanol (IPA) were used as the scavengers for $\rm h^+$ (ref. 54) and 'OH,55 respectively. Fig. 10 shows the removal efficiency of RhB by 54 wt% BiOI/KTaO₃ with and without the addition of various scavengers. Obviously, the degradation efficiency of dyes had no apparent change after the introduction of IPA, implying that 'OH played a negligible part in the photocatalytic process. However, with the introduction of TEOA or BQ into the photocatalytic system, the RhB degradation efficiency greatly decreased, and only 20.0% or 50.2% RhB was removed after 40 min of irradiation. These results revealed that 'O₂^{$-$} and h⁺ were the primary reactive species in the photodegradation of RhB.

Fig. 9 (a) Cycle experiment of degradation rate of RhB by 54 wt% BiOI/KTaO₃ sample, (b) XRD pattern of 54 wt% BiOI/KTaO₃ composite before and after cycle experiment.

The photoluminescence (PL) spectroscopy is widely employed to study the separation efficiency of photo-induced carriers. In general, the lower PL intensity indicates the higher separation efficiency of photo-generated electron–hole pairs,⁵⁶ corresponding to the higher photocatalytic efficiency. The PL spectra of all the catalysts are shown in Fig. 11. Clearly, the pure samples showed higher PL intensity compared to the $BiOI/KTaO₃$ composites, indicating that the separation efficiency of carriers in the BiOI/KTaO₃ composites was higher than those of BiOI and KTaO₃. The result can be ascribed to the inner electric field established between $KTaO₃$ and BiOI, which was conducive to the separation of photo-induced carriers.

In order to further investigate the separation and transfer of photo-induced electrons and holes, the photoelectrochemical measurements were carried out. Fig. 12a shows the photocurrent–time curves of the BiOI, $KTaO₃$ and BiOI/KTaO₃ composites under visible light irradiation. The photocurrent intensity showed the following orders: 54 wt% BiOI/KTaO₃ > 70 wt% BiOI/KTaO₃ > 37 wt% BiOI/KTaO₃ > BiOI > KTaO₃, which was in agreement with the degradation profiles. In general, the higher photocurrent implies the more effective separation of the photo-induced carriers. Therefore, the construction of the BiOI/ $KTaO₃$ heterostructures was helpful for the separation and

migration of charge carriers. Furthermore, EIS measurement can also be used to demonstrate the abovementioned results. From Fig. 12b, the resistance of pure BiOI and 54 wt% BiOI/ $KTaO₃$ was measured with and without visible-light irradiation, respectively. It can be seen that the radius of 54 wt% BiOI/ $KTaO₃$ composite was smaller than that of pure BiOI with or without irradiation, indicating that the transfer efficiency of the electrons in the 54 wt% $BiOI/KTaO₃$ composite was higher than that of BiOI. The results further confirmed that the p-n junction was helpful for the separation and migration of photo-induced carriers.

In order to confirm the semiconductor type of BiOI and $KTaO₃$, the Mott–Schottky test was performed. As shown in Fig. 13a, the slope of the Mott–Schottky curve for BiOI showed negative, suggesting that BiOI was a p-type semiconductor, and this result is in agreement with that obtained in a previous report.²³ Meanwhile, KTaO₃ was proved to be an n-type semiconductor with a positive slope of the Mott–Schottky curve shown in Fig. 13b. To determine the position of the valence band (E_{VB}) and conduction band (E_{CB}) potential of BiOI as well as KTaO₃, we followed the equations: $E_{VB} = X - E_0 + 0.5E_g$, $E_{CB} =$ E_{VB} – E_{g} , where E_{VB} and E_{CB} represent valence band and

Fig. 10 Effect of scavengers on the photodegradation of RhB by 54 wt% BiOI/KTaO₃ under visible light irradiation. Fig. 11 Photoluminescence (PL) spectra of all samples

Fig. 12 (a) Photocurrent responses of all samples under visible light irradiation, (b) EIS plots of BiOI and 54 wt% BiOI/KTaO₃ with and without visible light irradiation.

Fig. 13 Mott–Schottky plots of pure BiOI (a) and pure $KTaO₃$ (b).

conduction band edge potentials, respectively; X denotes absolute electronegativity and was determined by the composition of atoms; and E_0 refers to the energy of free electrons on the hydrogen scale (about 4.5 eV vs. NHE). The X of BiOI was calculated to be 5.94 eV, while that of $KTaO₃$ was 5.42 eV. According to the above-mentioned formula, the E_{VB} and E_{CB}

values of BiOI were determined to be 2.32 and 0.56 eV, respectively, and those of $KTaO₃$ were estimated to be 2.79 and -0.96 eV, respectively.

As shown in Fig. 14a, for BiOI, the Fermi level was close to the valence band (VB), while for $KTaO₃$, it was close to the conduction band (CB). When $KTaO₃$ was contacted with BiOI,

Fig. 14 Schematic diagram of (a) the band energy of BiOI and KTaO₃ before contact and (b) the formation of p–n junction and the charge transfer and separation process under visible light irradiation.

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the electrons transferred to BiOI from $KTaO₃$ until the Fermi level got balanced. Hence, an inner electronic field was formed between $KTaO₃$ and BiOI because of the difference in potential. When the composites were irradiated with visible light, only BiOI could be excited; the electrons jumped from the VB into the CB of BiOI, leaving holes on the VB of BiOI. Subsequently, the electrons, with the help of an inner electric field, were forced to transfer to the CB of KTaO₃ (Fig. 14b), facilitating the separation and migration of photo-excited carriers. Therefore, the p– n heterojunction generated between BiOI and $KTaO₃$ improved the separation efficiency of the photo-excited carriers (Fig. 14), resulting in a higher photocatalytic efficiency.

4. Conclusions

In summary, p-n heterojunction $BiOI/KTaO₃$ composite was successfully prepared through depositing BiOI nanosheets on the surface of the KTaO₃ nanocubes. The BiOI/KTaO₃ composites exhibited an enhanced photocatalytic property toward the photodegradation of RhB and phenol under the irradiation of visible light. When the loading amount of BiOI was 54 wt%, the BiOI/KTaO₃ composites displayed the highest photocatalytic efficiency for the degradation of RhB and phenol. The improved photocatalytic performance could be ascribed to the successful construction of p -n junction between BiOI and KTaO₃, which facilitates the separation and migration of photo-induced charge carriers, as verified by the results of PL, photocurrent response and EIS. **PSC** Advances Articles. Power and interest Article is licensed on the $\frac{1}{2}$ Creative Commons and the Creative Commons are the composites were interested with high $\frac{1}{2}$ and $\frac{1}{2}$ and $\frac{1}{2}$ and $\frac{1}{2}$ a

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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