RSC Advances

PAPER

Cite this: RSC Adv., 2020, 10, 9745

Investigation into the synthesis conditions of $CuMoO₄$ by an in situ method and its photocatalytic properties under visible light irradiation†

We[n](http://orcid.org/0000-0002-9270-0922)cheng Tan^{ab} and Jingfei Luan D^{*ab}

A kind of molybdenum and copper compound, CuMoO4, was prepared by a hydrothermal method. The construction and photocatalytic properties of $CuMoO₄$ have been characterized by X-ray diffraction (XRD), scanning electron microscopy (SEM), X-ray photoelectron spectroscopy (XPS), Fourier transform infrared spectroscopy (FTIR), Raman spectroscopy, transmission electron microscopy (TEM), UV-visible spectrometry and comprehensive thermal analysis. XRD analysis showed that samples which were synthesized under different hydrothermal time conditions were consistent, but the crystallinities of the samples were different. In another situation, disparate hydrothermal temperatures during the synthesis of CuMoO₄ would lead to the appearance of different samples. The band gap of CuMoO₄ was estimated to be 1.97 eV. It could be found from the results that $CuMoO₄$ was an indirect band gap semiconductor by simulating its band structure. The photocatalytic activities of $CuMoO₄$ were studied by means of monitoring the abilities of these compounds to degrade rhodamine B or 1H-benzotriazole in liquid media under visible light irradiation. Under different synthesis conditions, the hydrothermal time for obtaining the optimal degradation efficiency was 10 h, and the hydrothermal temperature was 180 $^{\circ}$ C. The results showed that $CuMoO₄$ had excellent degradation performance for rhodamine B or 1Hbenzotriazole. CuMoO₄ showed excellent mineralization efficiency for rhodamine B compared with Ndoped TiO₂ based on the reduction of total organic carbon (TOC) during the photocatalytic process. The photocatalytic degradation rate of rhodamine B by CuMoO₄ was 1.39 times that by N-doped TiO₂, and the degradation rate of TOC by CuMoO₄ was 1.53 times that by N-doped TiO₂. Based on the intermediate products which were detected by liquid chromatography/mass spectrometry (LC/MS), the possible degradation pathway of rhodamine B was derived. **PAPER**
 IDENTIFY AND CHECK CONSULTER SUBSECTION CON

Received 16th January 2020 Accepted 17th February 2020

DOI: 10.1039/d0ra00496k

rsc.li/rsc-advances

1 Introduction

Nano-structured materials have attracted a lot of attention because the properties of nanomaterials depend not only on their composition, but also on their size, shape and so on.¹⁻⁵ In the family of inorganic materials, metal molybdate was one of the most significant compounds due to its special composition, shape and size. More and more attention has been paid to metal molybdate because of its extensive applications in plentiful fields, such as industrial catalysts, partial oxidation of hydrocarbons, precursor synthesis catalysts, humidity sensors, scintillator materials, and for its magnetic properties, electrochemical properties, photoluminescence and microwave

applications. Furthermore, its areas of application have been extended to the fields of biology, medicine and chemical industry.⁶–¹² In the past few decades, the transition metal molybdates AMoO₄ (A = Cu, Ni, Co, Fe and Zn) and $B_3M_0^2O_9$ (B $=$ Cu, Zn) have been widely researched because of their unique structural properties, electronic properties, magnetic properties and catalytic properties.¹³⁻¹⁵ Among transition metal molybdates, copper molybdate was a prospective partial oxidation catalyst for light alkanes, especially for propane and isobutylene.¹⁶ There were six known polymorphs of CuMoO₄, such as α -CuMoO₄ at low temperature and β -CuMoO₄ at high temperature. The room-temperature triclinic (alpha) polymorph was studied. In the thermal analysis results, a small signal of a phase transition could be seen on the DSC curve around 570 °C,¹⁷ which belonged to the room-temperature (α) \rightarrow hightemperature (β) phase transition of CuMoO₄.

The application of copper molybdate compounds can be discovered in some fields. It was expressed that copper molybdate compounds have quite a potential for reducing the combustion temperature of carbon black.¹⁸ Thermal expansion properties, semiconducting properties and electrical transport

a School of Physics, Changchun Normal University, Changchun, Jilin 130032, China. E-mail: chendibd71@126.com

b State Key Laboratory of Pollution Control and Resource Reuse, School of the Environment, Nanjing University, Nanjing 210093, China. E-mail: jfluan@nju.edu. cn; Tel: +86-199-5193-9498

[†] Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ra00496k

properties have been studied.¹⁹⁻²¹ G. Steiner and R. Salzer studied the relationship between the temperature of CuMoO₄ and its optical properties.²² At room temperature, the CuMoO₄ crystal showed a strongly green color. Afterwards, with an increase in temperature, the color of $CuMoO₄$ transformed to dark brown at around 400 °C. This change in CuMoO₄'s color was absolutely reversible and the crystal structure was not broken during the temperature cycle. $CuMoO₄$ showed the similar color change during the temperature cycle at an oxygen partial pressure of $P_{\text{O}_2} = 100 \text{ Pa}$ or $P_{\text{O}_2} = 2 \times 10^5 \text{ Pa}$.

In recent years, the application of molybdenum-based oxide materials in photocatalysis and lithium-ion batteries has been extensively explored.^{23,24} For instance, PbMoO₄ and CdMoO₄ showed good photodegradation performance for rhodamine $B^{25,26}$ CoMoO₄ and Mn₂Mo₃O₈ had higher reversible capacity and excellent cycle stability in lithium-ion batteries.^{27,28} In some aspects, spherical structural materials attracted more attention for the photocatalytic degradation of dyes, photocurrent response and lithium-ion batteries.^{29,30} Based on the above analysis, we strived to investigate whether highly monodispersed CuMoO4 particles had great application potential in numerous electroluminescent devices, such as photocatalysis, photocurrent and energy conversion.

In this paper, we studied the performance of $CuMoO₄$ and made some characterizations of some properties. As for photocatalytic applications, previous studies have shown that semiconductor compounds could break down the most persistent organic pollutants, such as coloring matter, detergents and volatile organic compounds under ultraviolet light irradiation.31,32 Rhodamine B was utilized as a model contaminant in this paper to evaluate the degradation activity by $CuMoO₄$ under visible light irradiation because of its wide use and difficult biodegradation. Moreover, the construction and photocatalytic characterization of $CuMoO₄$ were also investigated in detail.

2 Experimental

2.1 Synthesis of nanocatalysts

CuMoO₄ was synthesized by a hydrothermal method. CuCl₂- \cdot 2H₂O and (NH₄)₆Mo₇O₂₄ \cdot 4H₂O with a purity of 99.99% (Sinopharm Group Chemical Reagent Co., Ltd., Shanghai, China) were used as the original materials. 5 mmol of $CuCl₂·2H₂O$ was dissolved in 30 mL of deionized water, and at the same time 5/ 7 mmol of $(NH_4)_6M_2O_{24}$ 4H₂O was dissolved in 30 mL of deionized water. Subsequently, the dissolved $(NH_4)_6$ - $Mo₇O₂₄·4H₂O$ solution was poured into the CuCl₂ solution. The molar ratio of copper element to molybdenum element was 1 : 1 in all experiments. The mixed solution was stirred for 30 min at room temperature, and then it was transferred to a polytetra fluoroethylene liner and placed in an autoclave; finally, it reacted at 180 °C for 10 h (110 °C, 150 °C, 180 °C, 200 °C, 220 °C temperature gradient and 4 h, 8 h, 10 h, 12 h, 14 h time gradient experiments were done). After hydrothermal reaction, the sample was taken out and filtered through a 0.45 μ m filter membrane. Subsequently, the filtered powder sample was baked in an oven for 4 h to obtain the precursor to $CuMoO₄$. The precursor to $CuMoO₄$ was placed in a muffle furnace (KSL

1700X, Hefei Kejing Materials Technology Co., Ltd., Hefei, China) and calcined at 500 °C to obtain a CuMoO₄ sample.

2.2 Characterization of samples

The particle formation of $CuMoO₄$ was measured with a transmission electron microscope (TEM, Tecnai F20 S-Twin, FEI Corporation, Hillsboro, Oregon, USA) with a 200 kV operating voltage. The chemical ingredients of $CuMoO₄$ were determined with a scanning electron microscope (SEM) which was equipped with an X-ray energy dispersion spectrum (SEM-EDS, LEO 1530VP, LEO Corporation, Pegnitz, Germany) with a 20 kV operating voltage. The Cu²⁺ content, Mo⁶⁺ content and O²⁻ content of $CuMoO₄$ and the valence states of the above elements were analyzed by X-ray photoelectron spectroscopy (XPS, ESCALABMK-2, VG Scientific Ltd., East Grinstead, UK). The chemical ingredients within the depth profile of $CuMoO₄$ were examined by the argon ion denudation method when X-ray photoelectron spectroscopy was utilized. The crystalline phase of CuMoO₄ was analyzed with an X-ray diffractometer (XRD, $D/$ MAX-RB, Rigaku Corporation, Tokyo, Japan) with Cu Ka radiation ($\lambda = 0.154056$ nm), and the patterns were collected at 295 K with a step-scan procedure in the range of $2\theta = 10$ –70°; at the same time, the step interval was 0.02° , and the time per step was 1 s. Finally, the accelerating voltage and applied current were 40 kV and 40 mA, respectively. Fourier transform infrared spectroscopy (FTIR, Nexus, Nicolet Corporation, Madison, Wisconsin, USA) was applied to examine the FTIR spectra of CuMoO₄; moreover, its spectral range was between 4000 and 400 cm^{-1} and the resolution was better than 0.09 cm^{-1} . In addition to the Fourier transform infrared study, Raman spectroscopy (Raman, LSM 710, Zeiss, German) was used to further investigate the molecular vibration and make a judgment about the functional groups of CuMoO4. Differential thermal analysis (DTA) was performed on a thermal synthesis analyzer (DTA-TG, Netzsch STA449C, NETZSCH Grinding & Dispersing Corporation, German) from 25 \degree C to 1000 \degree C at a linear heating rate of 10 °C min^{-1} . RSC Advances Article and the membrican state of the state is like the state of

2.3 Photoelectric response experiments

The UV-visible diffuse reflectance spectra of $CuMoO₄$ were measured with a Shimadzu UV-2550 UV-visible spectrometer (Shimadzu, Santa Clara, California, USA), and Ba SO_4 was utilized as the reference material. Photoluminescence spectroscopy (Fluoromax-4, HORIBA, Japan) was utilized to test the recombination rate of photo-induced carriers derived from the sample. The excitation wavelength was 300 nm and the slit width was 1 nm.

The photocatalytic activities of $CuMoO₄$ were assessed with rhodamine B $(C_{28}H_{31}C\text{N}_2O_3)$ (Tianjin Bodi Chemical Co., Ltd., Tianjin, China) or 1H-benzotriazole $(C_6H_5N_3)$ as the model substance. The photoreaction was implemented in a photochemical reaction apparatus (Nanjing Xujiang Machine Plant, Nanjing, China). The inner structure of the reaction apparatus was as follows: the lamp was put into a hollow quartz hydrazine structure, and laid in the middle of the reactor. The recycling water which flowed through the reactor was kept at a near

constant reaction temperature (20 \degree C), and the solution was continuously stirred and aerated. Twelve holes were utilized for the insertion of quartz tubes which were evenly arranged around the lamp, and the distances between the lamp and each hole were equal. The photocatalyst within the rhodamine B solution or 1H-benzotriazole solution was in a state of suspension under the condition of magnetic stirring. In this paper, the photocatalytic degradation of rhodamine B or 1H-benzotriazole was carried out with 25 mg of CuMoO₄ in 50 mL and 20 mg L⁻¹ pollutant aqueous solutions in quartz tubes with a 500 W xenon lamp (the main range of wavelength was from 420 nm to 800 nm) as the visible light source. Prior to visible light irradiation, the suspensions which contained the catalyst and pollutant solution were magnetically stirred in darkness for 30 min to ensure the establishment of an adsorption/desorption equilibrium among CuMoO4, the pollutant and atmospheric oxygen. During visible light illumination, the suspension was stirred at 500 r s^{-1} . The above experiments were carried out under conditions of oxygen saturation ([O₂]_{sat} = 1.02×10^{-3} mol L⁻¹). One of the quartz tubes was taken out of the photochemical reaction apparatus at various time intervals. The suspension

was filtered through $0.22 \mu m$ membrane filters. The rhodamine B filtrate was subsequently analyzed with a Shimadzu UV-2450 spectrometer (Shimadzu, Santa Clara, California, USA) with a detection wavelength of 554 nm. The $1H$ -benzotriazole filtrate was analyzed by liquid chromatography (LC, Dionex UltiMate 3000, Dionex, USA) with an EC-C18 column (5 μ m, 4.6 mm \times 250 mm). The mobile phase was a mixed solution of ultrapure water (A) and methyl alcohol (B) with a flow rate of 0.8 mL min⁻¹ and an injection volume of 20 μ L. The linear gradient elution method was determined as follows: the initial 90% A was reduced to 10% A within 20 min. Then the mobile phase was 100% B for 10 min. The column temperature was 303 K (30 \degree C).

The intermediate degradation products of rhodamine B were monitored with a liquid chromatography/mass spectrometer (LC/MS, Agilent 6530 Accurate-Mass Q-TOF LC/MS, Agilent, USA) with an EC-C18 column (2.7 μ m, 4.6 mm \times 50 mm) and a mode of positive electrospray ionization (ESI+). The mobile phase was a mixed solution of ultrapure water and methyl alcohol with a flow rate of 0.4 mL min⁻¹ and the injection volume was $4 \mu L$. The linear gradient elution method was determined as follows: the initial 90% ultrapure water was

Fig. 1 The XRD patterns of (a) $Cu_3(M_0O_4)_2(OH)_2$ and $NH_3(M_0O_3)_3$, and (b) $CuMoO_4$.

Fig. 2 The XRD patterns of (a) the CuMoO₄ precursor at different hydrothermal times and (b) CuMoO₄ at different hydrothermal times.

Fig. 3 SEM images of $CuMoO₄$ at different hydrothermal times.

reduced to 10% ultrapure water within 13 min. Subsequently the mobile phase was 100% methyl alcohol for 5 min and the column temperature was 303 K (30 $^{\circ}$ C).

3 Results and discussion

3.1 Powder synthesis

The phase structure and crystal structure of the as-prepared samples were further analyzed by the XRD method. The XRD patterns as well as the PDF standard cards of $CuMoO₄$ together with its precursors are shown in Fig. 1. As can be seen from Fig. 1, the sample $CuMoO₄$ and its precursors were consistent with the corresponding PDF card, indicating that the samples were successfully synthesized. Among them, the precursors which generated CuMoO₄ corresponded to $\text{Cu}_3(\text{MoO}_4)_2(\text{OH})_2$ and NH₃(MoO₃)₃. After calcination at 500 °C, the prepared sample $CuMoO₄$ was obtained from the precursors. The following reaction occurred during the calcination process:

$$
3Cu3(MoO4)2(OH)2 + NH3(MoO3)3 \rightarrow 9CuMoO4+ NH3† + 3H2O
$$
 (1)

Fig. 2 shows the XRD patterns of $CuMoO₄$ and its precursors under different conditions of hydrothermal time. Fig. 4 shows the SEM images of $CuMoO₄$ at different hydrothermal times. Firstly, under different hydrothermal time conditions, CuMoO₄ was consistent with the corresponding standard cards, indicating that the synthesized products were independent of the hydrothermal time. However, the intensity of the XRD peaks for CuMoO4 varied with different hydrothermal times. The XRD peak intensity of $CuMoO₄$ was the highest under hydrothermal conditions for 10 h. And at this point the performance of the precursors was consistent with the results of the product CuMoO4. The intensity of the XRD peaks indicated that the crystallinity was good to some extent, which was important for the catalytic performance of the sample. Furthermore, the exposed crystal faces of the main peak for $CuMoO₄$ were (101),

 (201) and (-202) . Optimization of these exposed crystal faces was one method for improving the catalytic performance of the photocatalysts.

It can be seen from Fig. 3 that $CuMoO₄$ was granulated and aggregated into a mass. When the hydrothermal time increased, the particles of $CuMoO₄$ became more and more round.

Fig. 4 (a) XRD patterns of $CuMoO₄$ at different hydrothermal temperatures and (b) an SEM image of CuMoO₄ at 200 °C.

Fig. 5 Comprehensive thermal analysis of the CuMo O_4 precursor.

Fig. 4(a) shows the XRD patterns of $CuMoO₄$ at different hydrothermal temperatures and Fig. 4(b) shows the SEM image of CuMoO₄ at 200 °C. In Section 2.1, we demonstrated the ability to synthesize $CuMoO₄$ with copper chloride and ammonium heptamolybdate. However, in the next experiment we found an interesting phenomenon which was that $CuCl₂·2H₂O$ and $(NH_4)_6Mo_7O_{24} \cdot 4H_2O$ could be used to obtain $Cu_3Mo_2O_9$ under certain conditions, and the above phenomenon was caused by temperature, as shown in Fig. 4(a). When the hydrothermal temperature was lowered below 150 \degree C, the substance which was synthesized with $CuCl₂·2H₂O$ and $(NH_4)_6Mo_7O_{24}$ 4H₂O did not match with the X-ray diffraction standard card of $CuMoO₄$, but matched the X-ray diffraction standard card of Cu₃Mo₂O₉. The reason was that CuCl₂ · $2H_2O$ and $(NH_4)_6Mo_7O_{24} \cdot 4H_2O$ could not form $NH_3(M_0O_3)_3$ under a hydrothermal temperature of 150 °C, and only $Cu_3(M_0O_4)_2($ $OH)$ ₂ was formed in the PTFE lining; as a result, a final product of $Cu₃Mo₂O₉$ was produced. In other words, only when the hydrothermal temperature reached 180 °C or higher, could $NH₃(MoO₃)₃$ be generated under high-temperature and high-

pressure conditions; as a result, the final product $CuMoO₄$ could be produced. And when the hydrothermal temperature was raised to 200 °C, it can be clearly seen from Fig. 4(b) that the particles of $CuMoO₄$ became rounded and tiny. The above analysis also revealed that the increase in temperature was beneficial for the successful synthesis of the sample under critical hydrothermal conditions. Fig. S1† shows a schematic diagram of the synthesis route for the product $CuMoO₄$. A comprehensive thermal analysis of $CuMoO₄$'s precursors is shown in Fig. 5. Combining the above analysis with the comprehensive thermal analysis which was shown in Fig. 5, the synthetic pathways of $CuMoO₄$ could be roughly inferred, as shown in Fig. S1.†

As for CuMoO₄'s precursor which can be found in Fig. 5, the TG-DTA curve can also be divided into two stages. In the first stage, CuMoO₄'s precursor lost a part of the weight in the range 200–412 \degree C, and the DTA curve in this temperature range had an exothermic peak. This was due to a thermal decomposition reaction which occurred in this process: $3Cu_3(M_0O_4)_2(OH)_2$ + $NH₃(MoO₃)₃ = 9CuMoO₄ + NH₃ + 3H₂O$. In the second stage, new-phase $CuMoO₄$ had been losing weight in the range 687– 1000 \degree C, and at this time DTA had a sharp endothermic peak, which was caused by the melting of $CuMoO₄$ and volatilization of CuMoO₄. The total weight loss ratio of CuMoO₄'s precursor in the range 0-1000 $^{\circ}$ C was 20.32%.

3.2 Characterization of CuMoO₄

Fig. 6 shows a TEM image of $CuMoO₄$ as well as the selectedarea electron diffraction (SAED) pattern and lattice fringe of $CuMoO₄$. It can clearly be seen that the microstructure and the interface properties of the $CuMoO₄$ powders were further characterized. Fig. $6(a)$ shows that CuMoO₄ was granular, and the above results are consistent with the SEM results for CuMoO4. The inset within Fig. 6(b) presents the selected-area electron diffraction pattern, which can further confirm that the sample $CuMoO₄$ was single crystalline. The selected-area electron diffraction pattern of $CuMoO₄$ conformed to the triclinic system, and the crystal plane in the selected-area electron diffraction pattern had (201) and (101) faces, and the lattice stripe pitch was 0.373 nm, which corresponded to the (201)

Fig. 6 (a) A TEM image of CuMoO₄, and (b) an SAED pattern of CuMoO₄ and the lattice fringe of CuMoO₄.

crystal plane. This result was consistent with the crystal faces which corresponded to the XRD peaks of CuMoO₄.

Fig. 7 shows the FTIR spectrum of $CuMoO₄$. The FTIR spectrum analysis of $CuMoO₄$ particles was investigated in detail. According to Fig. 7, we can see that the absorption bands of CuMoO4 that was prepared by a hydrothermal method were mainly concentrated within 1000 $\mathrm{cm}^{-1}.$ The absorption peaks at 480–510 cm⁻¹ for CuMoO₄ were due to Cu-O bonds. In combination with the Raman spectrum (Fig. 8), the peaks at 796–937 cm^{-1} might correspond to the stretching vibration modes of the $MoO₄$ tetrahedron. For a tetrahedral ion such as MoO4, the asymmetric Mo–O stretching is intensive in the FTIR spectrum, and weak in the Raman spectrum. In contrast, the symmetric Mo–O stretching is very weak in the FTIR spectrum and very intense in the Raman spectrum. This meant that the band at 966 cm⁻¹ (Fig. 8) might only be the symmetric Mo-O stretching (strongest in the Raman spectrum, and weak in the FTIR spectrum). Similarly, the other three bands at 796– 937 cm^{-1} (weak in the Raman spectrum, strong in the FTIR

Fig. 8 The Raman spectrum of $CuMoO₄$

spectrum) belong to the three components of the Mo–O asymmetric stretching mode (F2, triple degenerated).¹⁷

The Raman spectrum was used to further examine the structural properties of the prepared photocatalysts. The Raman spectrum of $CuMoO₄$ is shown in Fig. 8. It can be observed from the Raman spectrum of $CuMoO₄$ that the mode around 336 cm^{-1} is the bending vibration which corresponds to CuO.^{33,34} The stretching vibration modes of the MoO₄ tetrahedron were observed at around 807 cm^{-1} , 841 cm^{-1} , 935 cm^{-1} and 966 cm^{-1} .³⁵⁻³⁷ The characteristic peak around 935 cm^{-1} corresponded to the symmetric stretching vibration peak of $Mo=O³⁸$ Based on the above analysis, we can deduce that the characteristic peaks at 807 cm^{-1} , 841 cm^{-1} and 935 cm^{-1} for CuMoO4 correspond to the stretching vibration modes of the $MoO₄ tetrahedron. According to Berrelough,³⁹ the peak around$ 887 cm^{-1} for CuMoO₄ corresponded to the Mo-O $(3,3^{\prime})$ vibration.

In order to gain further insight into the interfacial electronic structures, we used the XPS technology to investigate the surface electronic structure and bonding configuration of CuMoO₄. The survey XPS spectra of CuMoO₄ are shown in Fig. 9(a–d). The peaks which originated from Fig. 9 corresponded to the C element, Cu element, Mo element and O element, which were also observed in the survey XPS spectra of CuMoO4. In the survey XPS spectra, the peak at 284.0 eV was attributable to the adventitious carbon in CuMo $O₄$, and the adventitious carbon was determined to be standard carbon. In the O 1s spectrum of $CuMoO₄$, the asymmetric peak at 530.8 eV was assigned to the binding energy of O^{2-} species characteristic of oxide materials.⁴⁰ The shoulder peak which was observed at 530.2 eV could be attributed to the chemisorbed oxygen species on the surface of CuMoO₄. In the fitted peaks of Cu 2p in CuMoO₄, a major peak at 934.8 eV with a shoulder at 932.1 eV was ascribed to the $2p_{3/2}$ core level of Cu²⁺. The binding energy of Mo 3d of CuMoO₄ was 232.7 eV or 235.8 eV.

Table 1 provides the experimental binding energies of the characteristic peaks of all elements which exist in $CuMoO₄$ and the binding energy information after C correction. The distance of the Mo element peaks was 3.1 eV, as shown in Fig. 9(d). By comparing the XPS standard binding energy data which is shown in Table 1 and the chemical shifts of each element, the valence state of each element in $CuMoO₄$ was determined, and the results showed that the valence states of Cu, Mo or O are +2, +6 or -2 .

3.3 Photocatalytic performance characterization of CuMoO₄

Fig. 10 shows the diffuse reflection spectra of $CuMoO₄$. The optical absorption properties of the as-synthesized samples were studied by UV-vis diffuse reflectance spectroscopy, as shown in Fig. 10.

The absorption spectrum of $CuMoO₄$ can be found in Fig. 10. For a crystalline semiconductor, the optical absorption near the band edge of the crystalline semiconductor follows eqn (2) :⁴¹

$$
\alpha h\nu = A \times (h\nu - E_{g})^{n} \tag{2}
$$

$$
E_{\rm g} = 1240/\lambda \tag{3}
$$

Fig. 9 CuMoO₄, Cu element, Mo element and O element XPS spectra: (a) full XPS spectra of CuMoO₄, (b) XPS spectrum of O 1s, (c) XPS spectrum of Cu 2p and (d) XPS spectrum of Mo 3d.

where A , α , E_g , ν and λ are the proportional constant, absorption coefficient, band gap, light frequency and the wavelength which correspond to the absorption edge, respectively. In the above equation, n determines the character of the transition in a semiconductor. $E_{\rm g}$ and *n* could be calculated by the following steps: (i) plotting $ln(\alpha h\nu)$ versus $ln(h\nu - E_{\rm g})$ by assuming an approximate value of $E_{\rm g}$, which can be calculated from eqn (3); (ii) deducing the value of *n*; and (iii) refining the value of E_g . From Fig. 10, we find that the absorption edge of $CuMoO₄$ is about 539 nm, indicating that the estimated E_g of CuMoO₄ was 2.30 eV. Subsequently, from the plot of $ln(\alpha h\nu)$ versus $ln(h\nu E_g$), we find that the slope of the line part was about 1.01. Therefore, the n which derived from CuMoO₄ was 2. After plotting $(\alpha h\nu)^2$ versus hv and extrapolating the plot to $(\alpha h\nu)^2 = 0$, the accurate value of E_g for CuMoO₄ was calculated to be 1.97 eV. The above results indicated that the optical transition

for CuMoO4 was indirectly allowed, and, at the same time, $CuMoO₄$ possessed a narrow band gap.

Fig. 11(a) shows the ultraviolet photoelectron spectroscopy (UPS) spectra of CuMoO4. According to the UPS results, the value of the valence band of $CuMoO₄$ was calculated to be 8.65 eV which was related to the Fermi energy. Combined with the band gap, the value of the conduction band of $CuMoO₄$ was found to be 10.62 eV. The band structure of $CuMoO₄$ is shown in Fig. 11(b).

Fig. 12 presents the photoluminescence (PL) spectra of both N-doped TiO₂ and CuMoO₄ photocatalysts with an excitation wavelength of 300 nm. $CuMoO₄$ showed a sharp emission peak at 355 nm. The PL spectra revealed that the emission of CuMoO4 was strongly dependent on its crystallinity. Benchikhi et al. attributed the PL emission of $CuMoO₄$ to the intrinsic slightly distorted $\text{[MoO}_{4}^{2-}\text{]}$ tetrahedra.⁴² On the other hand, PL

Fig. 10 The diffuse reflection spectrum of CuMoO₄. Insets: (a) plot of ln($\alpha h\nu$) versus ln(h $\nu-E_g$) and (b) plot of ($\alpha h\nu$)^{1/2} versus h ν .

Fig. 11 (a) UPS spectrum of CuMoO₄, and (b) the band structure of CuMoO₄, which was related to the Fermi energy

spectroscopy is a method for exploring both the transfer process of the charge carrier on the interface of photocatalysts and the recombination rate involved with the photogenerated electron– hole pairs. Herein, the PL spectra were employed to reveal the recombination abilities of N-doped TiO₂ or CuMoO₄. The intensity of the PL spectra for N-doped $TiO₂$ was very much higher than that for CuMoO₄. This showed that CuMoO₄ had a lower recombination rate of photogenerated electron–hole pairs, which was more conducive to the progress of photocatalytic reactions.

In order to further explore the photocatalytic properties of CuMoO₄, the band structure of CuMoO₄ was calculated. According to the DFT calculations, the band structure of CuMoO4 is shown in Fig. 13. For the band structure diagram,

the k -points path G-F-Q-Z-G is presented in detail.⁴³ The top of the valence band lay between the F point and the Q point, and the bottom of the conduction band was located between the Q point and the Z point. It was found that $CuMoO₄$ was an indirect band gap semiconductor; moreover, the valence band maximum and conduction band minimum of the calculated CuMoO₄ were not located at the same k -point. Therefore, in this case, the value of n in eqn (2) was 2, which was consistent with the results of the diffuse reflectance spectrum of $CuMoO₄$. In contrast, the calculated band gap of $CuMoO₄$ was only 1.12 eV. The reason was that the description of exchange–correlation effects utilized the general gradient approximation (GGA), which always leads to an under-estimation of band gaps for

Fig. 12 Photoluminescence spectra of $CuMoO₄$ and N-doped TiO₂.

transition metals, resulting in a difference from the experimental result.^{44,45}

3.4 Photocatalytic properties of CuMoO4

The progress of photocatalysis with the semiconductor compound as catalyst could be described briefly as follows.^{46,47} Firstly, the semiconductor compound absorbs photons. As a result, photoinduced electron–hole pairs are generated within the semiconductor compound particles. Subsequently, the diffusion of the charge carriers to the surface of the semiconductor compound particle would follow. At the same time, the active sites of the surface of the semiconductor compound particles might adsorb a lot of organic pollutant particles. Finally, the decomposition of the organic pollutants would be performed by charge carriers.

Fig. 14 presents the changes in the UV-vis spectra of rhodamine B under visible light irradiation ($\lambda > 420$ nm) in the presence of CuMoO4. The above measurements were performed under conditions of oxygen saturation $([O_2]_{sat} = 1.02 \times$

Fig. 14 Spectral changes of aqueous solutions of rhodamine B due to visible light irradiation in the presence of CuMoO₄.

 10^{-3} mol L^{-1}). It can clearly be seen from Fig. 14 that the typical rhodamine B peak is at 554 nm. An obvious reduction in absorbance could be observed within 300 minutes.

Fig. 15 shows the photocatalytic removal efficiency of rhodamine B under visible light irradiation for 120 min with CuMoO4 as photocatalyst using different hydrothermal times or temperatures. It can be seen from Fig. $15(a)$ that the CuMoO₄-10 h group had the highest degradation efficiency for rhodamine B, and the degradation rate reached 30% within 120 min. The lowest degradation efficiency was with the CuMoO₄-14 h group. Anyway, $CuMoO₄$ still had a degradation effect on rhodamine B at different hydrothermal times under visible light irradiation. Fig. 15(b) shows the degradation of rhodamine B by the products which were synthesized at different hydrothermal temperatures. At a hydrothermal temperature of 110 °C or 150 \degree C, the sample had almost no degradation efficiency for rhodamine B. This was due to the product which was synthesized being $Cu₃Mo₂O₉$ at this time instead of CuMoO₄, which was consistent with the results of the previous analysis. The highest degradation efficiency came from $CuMoO₄$ which was synthesized at a hydrothermal temperature of 180 °C.

Fig. 16 and 17 show the photocatalytic degradation rate for rhodamine B or 1H-benzotriazole under visible light irradiation in the presence of $CuMoO₄$. When rhodamine B was used as a contaminant, the photodegradation rate was 29.12% after visible light irradiation for 120 minutes, while the photodegradation rate for 1H-benzotriazole was 71.13% by CuMoO₄. Based on the absorbance changes of rhodamine B with light irradiation time, the kinetic curves of rhodamine B degradation under visible light irradiation were figured out (Fig. $S2(a)$ †). The first-order rate constant for $CuMoO₄$ was estimated to be 0.00297 min⁻¹ or 0.00933 min⁻¹ with rhodamine B or 1H-benzotriazole as contaminant, respectively. This fact indicated that CuMoO₄ was more efficient at degrading $1H$ -benzotriazole than rhodamine B under visible light irradiation.

Fig. 17 shows the change in total organic carbon (TOC) during the photocatalytic degradation of rhodamine B or 1H-

Fig. 15 The photodegradation of rhodamine B with (a) different hydrothermal times and (b) different hydrothermal temperatures in the presence of CuMoO4.

Fig. 16 The photodegradation of rhodamine B and 1H-benzotriazole under visible light irradiation in the presence of CuMoO₄.

benzotriazole under visible light irradiation in the presence of CuMoO4. TOC was measured using a TOC detector (TOC, vario TOC, Elementar, Germany). It has a detection limit of 2 μ g L⁻¹ and an accuracy of RSD $(C) < 1\%$. The TOC measurements revealed the disappearance of organic carbon when the IC solution which contained $CuMoO₄$ or N-TiO₂ was exposed to visible light irradiation. The results showed that 27.38% of the TOC decrease was obtained after visible light irradiation for 120 minutes when rhodamine B was degraded, while 63.58% of TOC had decreased when 1H-benzotriazole was degraded. The apparent first-order rate constant k was estimated to be 0.00268 or 0.00852 min⁻¹ with rhodamine B or 1H-benzotriazole as the contaminant, respectively (Fig. S2(b)†).

For further comparison, Fig. 18 and 19 depict the concentration of rhodamine B and the total organic carbon (TOC) concentration changes with $CuMoO₄$ or N-doped TiO₂ as photocatalyst under visible light irradiation, respectively. When N- $TiO₂$ was utilized as a catalyst, the photodegradation rate of rhodamine B was 59.12% after visible light irradiation for 300

Fig. 17 TOC plots during the photocatalytic degradation of rhodamine B and 1H-benzotriazole under visible light irradiation in the presence of CuMoO₄

minutes, while the photodegradation rate of rhodamine B was 82.2% by $CuMoO₄$. The results showed that the photodegradation rate of rhodamine B with $CuMoO₄$ as a catalyst was higher than that with N-TiO₂ as a catalyst. In addition, based on the absorbance changes of rhodamine B with light irradiation time, the kinetic curves of rhodamine B degradation under visible light irradiation were figured out (Fig. $S3(a)$ †). The above results demonstrated that the photocatalytic kinetics of rhodamine B degradation with CuMoO₄ or N-TiO₂ as photocatalyst followed a first-order nature. The first-order rate constant for rhodamine B degradation was estimated to be 0.006 min⁻¹ or 0.00285 min⁻¹ with CuMoO₄ or N-TiO₂ as catalyst, respectively. This fact indicated that $CuMoO₄$ was more efficient than N-TiO₂ for the photocatalytic degradation of rhodamine B under visible light irradiation.

Fig. 19 shows the change in total organic carbon (TOC) during the photocatalytic degradation of rhodamine B with

Fig. 18 The photocatalytic degradation of rhodamine B under visible light irradiation with CuMoO₄ and N-doped TiO₂ as the photocatalyst, respectively.

Fig. 19 TOC plots during the photocatalytic degradation of rhodamine B under visible light irradiation with $CuMoO₄$ and N-doped TiO₂ as the photocatalyst, respectively.

CuMoO₄ or N-TiO₂ as catalyst under visible light irradiation. The results showed that 53.21% of the TOC decrease was obtained after visible light irradiation for 300 minutes when N- $TiO₂$ was utilized as the photocatalyst, while 81.16% of TOC decreased with $CuMoO₄$. The apparent first-order rate constant k was estimated to be 0.00534 or 0.00245 min⁻¹ with CuMoO₄ or N-TiO₂ as the photocatalyst, respectively (Fig. S3(b)[†]).

In order to probe the stability of the catalyst $CuMoO₄$, we carried out an experiment of the repeated degradation of rhodamine B. Fig. 20 shows the experimental results of the repeated degradation of rhodamine B under the same experimental conditions after four cycles of $CuMoO₄$ recovery. The results showed that the removal rates of rhodamine B showed little change after four reuses of $CuMoO₄$. It could be inferred

Fig. 20 Repeated photocatalytic degradation tests of rhodamine B with $CuMoO₄$ as the photocatalyst.

that the catalytic performance of $CuMoO₄$ was relatively stable and repeatable.

For the sake of a detailed analysis of the photocatalytic degradation of rhodamine B, the chromatograms and mass spectra of degraded rhodamine B (taken after 0 min and 300 min of visible light irradiation with $CuMoO₄$ as photocatalyst) were analyzed by LC/MS, as shown in Fig. 21 and S4.† It can be seen from the chromatogram that the retention time of rhodamine B was about 12 min. When the photocatalytic reaction proceeded, the peak intensity of rhodamine B was constantly decreasing, and it became extremely low at 300 min. The peaks of various intermediate products also appeared. The above results further indicated that rhodamine B was continuously degraded. By comparing the mass-to-charge ratio (m/z) , the intermediate products which may be formed by the photocatalytic degradation of rhodamine B with $CuMoO₄$ as photocatalyst are listed in Table 2.

By comparing the mass-to-charge ratio of the intermediate products, it was found that the II–VI of Fig. S4† corresponded to one de-ethylated product, two de-ethylated products, three deethylated products and four de-ethylated products of rhodamine B, wherein IV was the isomer of III, which corresponded to the intermediate product that was formed after two ethyl groups were removed from rhodamine B. One structural type of rhodamine B molecule had one ethyl group which was removed from each side, and the other structural type had two ethyl groups removed from one side of rhodamine B. The above results all indicated that rhodamine B was in the process of being gradually degraded under visible light irradiation. The conjectured degradation pathway of rhodamine B is shown in Fig. 22.

In order to verify the major reactive species inducing the degradation of rhodamine B with CuMo $O₄$ as catalyst under visible light irradiation, dissociative scavenger experiments were carried out by adding different scavengers into the system. The photodegradation of rhodamine B over $CuMoO₄$ in the face

Fig. 21 Chromatograms of non-degraded (0 min) and degraded rhodamine B (300 min) during visible light irradiation with CuMoO₄ as the photocatalyst.

of various scavengers is expounded in Fig. 23. During the experiment, BQ (benzoquinone) and IPA (isopropanol) were used as scavengers for 'O $_2^-$ and 'OH, respectively. The amount of BQ or IPA added was set to 1 mL and the concentration of BQ or IPA was set to 10 mmol L^{-1} . As shown in Fig. 23, after the addition of BQ or IPA, the photocatalytic activity showed an obvious reduction. BQ had stronger inhibitory effect on photocatalytic activity than IPA. The above results suggested that the photoreaction process was dominated mainly by $^{\cdot}O_2^{\cdot-}$ and 'OH had a secondary effect on the photoreaction process. Additionally, the possible active radicals in the process of photocatalytically degrading rhodamine B were measured by the EPR characterization method, as shown in Fig. S5.† This showed a super response to $\mathrm{`O_{2}^-}$ and a certain response to 'OH. Based on the above results, a possible mechanism for the charge separation and photocatalytic reaction for $CuMoO₄$ was inferred:

$$
CuMoO4 + h\nu \to h+ + e-
$$
 (4)

$$
e^- + O_2 \rightarrow 'O_2^-
$$
 (5)

$$
(H_2O \rightleftharpoons H^+ + OH^-) + h^+ \rightarrow H^+ + 'OH \tag{6}
$$

Rhodamine $B + 'O_2^-/'OH/h^+$ \rightarrow intermediates \rightarrow degradation products (7)

In the first step, photogenerated holes $(h⁺)$ and photogenerated electrons (e^{-}) came into being on the surface of CuMoO₄ particles (eqn (4)). Many published works⁴⁸⁻⁵⁰ have confirmed that two oxidative agents could mainly be concerned under visible light irradiation: O_2 ⁻ radicals and 'OH radicals. Subsequently, photogenerated electrons and photogenerated holes reacted with water and dissolved oxygen to form ${^{\cdot}O_2}^$ radicals and 'OH radicals (eqn (5) and (6)). Then the free radicals in the solution reacted with rhodamine B to form intermediate products, which were eventually degraded into small

small molecule organic matters

Fig. 22 A possible photodegradation pathway of rhodamine B in water.

Fig. 23 The photodegradation of rhodamine B over $CuMoO₄$ without a scavenger and in the presence of BQ and IPA.

molecular substances (eqn (7)). In the photocatalytic process, O_2 ⁻ was the dominant active radical.

4 Conclusions

In summary, the synthetic route of $CuMoO₄$ was systematically studied. When copper chloride and ammonium molybdate were utilized as raw materials, the obtained product changed with variations in temperature. When the hydrothermal temperature was lower than 150 \degree C, the obtained product was $Cu₃Mo₂O₉$. When the hydrothermal temperature was above 180 °C, CuMoO₄ was obtained. The reason for the above results was that copper chloride and ammonium molybdate reacted during the hydrothermal process at a hydrothermal temperature below 150 °C to form a precursor, $\text{Cu}_3(\text{MoO}_4)_2(\text{OH})_2$, which was dehydrated to become $Cu₃Mo₂O₉$ by calcination at 500 °C. Copper chloride and ammonium molybdate reacted at 180 \degree C or higher to form $Cu_3(M_0O_4)_2(OH)_2$ and $NH_3(M_0O_3)_3$, which were chemically mixed after calcination at 500 °C to form CuMoO₄. The reaction path was obtained by comprehensive thermal analysis. The XPS results for $CuMoO₄$ indicated that the chemical valence values of Cu, Mo and O were $+2$, $+6$ and -2 . The band gap of $CuMoO₄$ was measured as 1.97 eV using a UVvisible diffuse reflectance spectrometer. The degradation of rhodamine B with CuMoO₄ as catalyst under different synthesis conditions was investigated. Under different synthesis conditions, the hydrothermal time for the optimal degradation efficiency was 10 h, and the hydrothermal temperature for the optimal degradation efficiency was 180 °C. In experiments degrading rhodamine B and $1H$ -benzotriazole with CuMoO₄, $CuMoO₄$ had a higher degradation efficiency for 1H-benzotriazole. The photocatalytic decomposition of rhodamine B aqueous solution was realized under visible light irradiation in the presence of $CuMoO₄$ or N-doped TiO₂. The results showed that CuMoO4 had higher photocatalytic activity than N-doped TiO2 for degrading rhodamine B under visible light irradiation. CuMo $O₄$ could degrade deep-red rhodamine B solution

into a nearly colorless solution after 300 minutes under visible light irradiation, demonstrating that $CuMoO₄$ showed high degradation activity. $CuMoO₄$ showed strong potential application prospects in the field of photocatalysis.

Author contributions

Jingfei Luan was involved with all aspects of the work, including visualizing, planning and data explication. Wencheng Tan carried out the experiments, analyzed the data and wrote the paper. All authors read and approved the manuscript.

Conflicts of interest

The authors declare no conflicts of interest.

Acknowledgements

This study was supported by a grant from the Research Foundation of Innovative Research and Supporting Program of Changchun Normal University in 2019, a grant from the Research Foundation of Science and Technology Bureau of Jilin Province for Middleaged Technological Innovative Scientist and Team Based on Environmental Function Material and Water Pollution Control, a grant from China–Israel Joint Research Program in Water Technology and Renewable Energy (Grant No. [2010]30), and the National Natural Science Foundation of China (No. 20877040).

References

- 1 B. I. Kharisov, H. V. R. Dias, O. V. Kharissova, et al., Cheminform, 2012, 2, 9325–9358.
- 2 L. Cui, J. Wu and H. Ju, Biosens. Bioelectron., 2015, 63, 276– 286.
- 3 X. Tan, Bioresour. Technol., 2016, 212, 318–333.
- 4 X. Gan, H. Zhao and X. Quan, Biosens. Bioelectron., 2017, 89, 56–71.
- 5 S. G. Ullattil, S. B. Narendranath, S. C. Pillai, et al., Chem. Eng. J., 2018, 343, 708–736.
- 6 S. M. Pourmortazavi, S. S. Hajimirsadeghi, M. Rahimi-Nasrabadi and M. M. Zahedi, Mater. Sci. Semicond. Process., 2013, 16, 131–137.
- 7 S. M. Pourmortazavi, S. S. Hajimirsadeghi, M. Rahimi-Nasrabadi and I. Kohsari, Synth. React. Inorg., Met.-Org., Nano-Met. Chem., 2012, 42, 746–751.
- 8 S. M. Pourmortazavi, S. S. Hajimirsadeghi, M. Rahimi-Nasrabadi and I. Kohsari, Chem. Eng. Commun., 2011, 198, 1182–1188.
- 9 R. Sundaram and K. S. Nagaraja, Sens. Actuators, B, 2004, 101, 353.
- 10 J. L. Brito, A. L. Barbosa, A. Albornoz and F. Severino, Catal. Lett., 1994, 26, 329.
- 11 J. H. Ryu, J. W. Yoon, C. S. Lim and K. B. Shim, Key Eng. Mater., 2006, 317, 223.
- 12 J. P. Liu, X. T. Huang, Y. Y. Li and Z. K. J. Li, J. Mater. Chem., 2007, 17, 2754.
- 13 L. A. Palacio, A. Echavarría, L. Sierra and E. A. Lombardo, Catal. Today, 2005, 107–108, 338.
- 14 J. H. Ryu, S. M. Koo, J. W. Yoon, C. S. Lim and K. B. Shim, Mater. Lett., 2006, 60, 1702.
- 15 J. S. Xu and D. F. Xue, J. Solid State Chem., 2007, 180, 119– 126.
- 16 M. Najafi, A. Abbasi, M. Masteri-Farahani and V. H. Nunes Rodrigues, Inorg. Chim. Acta, 2015, 433, 21.
- 17 E. Sajo, P. Bakos, M. Szilagyi, et al., Inorg. Chem., 2018, 57, 13679–13692.
- 18 M. A. Hasan, M. I. Zaki, K. Kumari and L. Pasupulety, Thermochim. Acta, 1998, 320, 23–32.
- 19 G. Mul, J. P. A. Neeft, F. Kapteijn, M. Makkee and J. A. Moulijn, Appl. Catal., B, 1995, 6, 339–352.
- 20 C. H. Wang and H. S. Wenig, Appl. Catal., A, 1998, 170, 73– 80.
- 21 J. Haber, T. Machej, L. Ungier and J. Ziolkowski, J. Solid State Chem., 1978, 25, 207–218.
- 22 G. Steiner, R. Salzer and W. Reichelt, Fresenius. J. Anal. Chem., 2001, 370, 731–734.
- 23 A. F. Gouveia, J. C. Sczancoski, M. M. Ferrer, A. S. Lima, M. R. M. C. Santos, M. S. Li, R. S. Santos, E. Longo and L. S. Cavalcante, Inorg. Chem., 2014, 53, 5589. Published on 11 L. A. Published on 12:30 Article is licensed on 12:21, 198, 2022 4:22, 2022 4:22, 2020. Downloaded on 2021/2024 4:12:49 AM. This article is licensed under a Creative Commons Articles. The Camera Creative Co
	- 24 Z. Xu, Z. Li, X. Tan, C. M. B. Holt, L. Zhang, B. S. Amirkhiz and D. Mitlin, RSC Adv., 2012, 2, 2753.
	- 25 W. S. Wang, L. Zhen, C. Y. Xu, W. Z. Shao and Z. L. Chen, CrystEngComm, 2013, 15, 8014.
	- 26 M. R. D. Bomio, R. L. Tranquilin, F. V. Motta, C. A. Paskocimas, R. M. Nascimento, L. Gracia, J. Andres and E. Longo, J. Phys. Chem. C, 2013, 117, 21382.
	- 27 C. T. Cherian, M. V. Reddy, S. C. Haur and B. V. R. Chowdari, ACS Appl. Mater. Interfaces, 2013, 5, 918.
	- 28 Y. M. Sun, X. L. Hu, W. Luo and Y. H. Huang, J. Mater. Chem., 2011, 21, 17229.
	- 29 D. D. Cai, L. X. Ding, S. Q. Wang, Z. Li, M. Zhu and H. H. Wang, Electrochim. Acta, 2014, 139, 96.
	- 30 D. Li, Q. Qin, X. C. Duan, J. Q. Yang, W. Guo and W. J. Zheng, ACS Appl. Mater. Interfaces, 2013, 5, 9095.
- 31 Y. Benshahar and U. Banin, Top. Curr. Chem., 2016, 374, 149–174.
- 32 M. Ahmad, E. Ahmed, Y. W. Zhang, et al., Curr. Appl. Phys., 2013, 13, 697–704.
- 33 L. Debbichi, M. C. Marco de Lucas, J. F. Pierson and P. Krüger, *J. Phys. Chem. C*, 2012, 116, 10232-10237.
- 34 G. Kliche and Z. V. Popovic, Phys. Rev. B: Condens. Matter Mater. Phys., 1990, 42, 10060–10066.
- 35 R. Nakajima, M. Abe, Y. Benino, T. Fujiwara, H. G. Kim and T. Komatsu, J. Non-Cryst. Solids, 2007, 353, 85–93.
- 36 M. Maczka, J. Solid State Chem., 1997, 129, 287–297.
- 37 L. Seguin, M. Figlarz, R. Cavagnat and J. C. Lassegues, Spectrochim. Acta, Part A, 1995, 51, 1323–1344.
- 38 C. E. Ciocan, E. Dumitriu, T. Cacciaguerra, F. Fajula and V. Hulea, Catal. Today, 2012, 198, 239–245.
- 39 C. Barraclough, G. J. Lewis and R. S. Nyholm, J. Chem. Soc., 1959, 3552–3555.
- 40 Y. Yao, F. Lu, Y. Zhu, F. Wei, X. Liu, C. Lian and S. Wang, J. Hazard. Mater., 2015, 297, 224–233.
- 41 J. Tauc, R. Grigorovici and A. Vancu, Phys. Status Solidi, 1966, 15, 627–637.
- 42 M. Benchikhi, R. E. Ouatib, et al., Int. J. Miner., Metall. Mater., 2016, 23, 1340–1345.
- 43 Z. Zhao, H. Wu, H. He, X. Xu and Y. Jin, Adv. Funct. Mater., 2014, 24, 4698–4705.
- 44 C. Xu, Y. Zhang, J. Chen, J. Lin, X. Zhang, Z. Wang and J. Zhou, Appl. Catal., B, 2017, 204, 324–334.
- 45 K. Yang, W. Q. Huang, L. Xu, K. W. Luo, Y. C. Yang and G. F. Huang, Mater. Sci. Semicond. Process., 2016, 41, 200– 208.
- 46 Y. J. Yang and J. F. Luan, Molecules, 2012, 17, 2752–2772.
- 47 M. J. Chang, H. Wang, H. L. Li, et al., J. Mater. Sci., 2018, 53, 3682–3691.
- 48 S. Q. Huang, Y. G. Xu, M. Xie, et al., Colloids Surf., A, 2015, 478, 71–80.
- 49 B. Subash, B. Krishnakumar, M. Swaminathan, et al., Langmuir, 2013, 29, 939–949.
- 50 H. Q. Sun, G. L. Zhou, S. Z. Liu, et al., Chem. Eng. J., 2013, 231(9), 18–25.