


Cite this: *RSC Adv.*, 2020, 10, 21887

# Piezoelectric polymer nanofibers for pressure sensors and their applications in human activity monitoring†

Minmin Zhu,<sup>a</sup> Soon Siang Chng,<sup>b</sup> Weifan Cai,<sup>b</sup> Chongyang Liu<sup>a</sup> and Zehui Du<sup>a</sup>

Miniaturized, wearable and self-powered sensors are crucial for applications in artificial intelligence, robotics, healthcare, and communication devices. In particular, piezoelectric polymer-based sensing systems have the advantages of light weight, large piezoelectricity and mechanical flexibility, offering great opportunities in flexible and stretchable electronic devices. Herein, free-standing large-size nanofiber (NF) membranes have been fabricated by an electrospinning technique. Our results show that the as-synthesized P(VDF-TrFE) NFs are pure  $\beta$ -phase and exhibit excellent mechanical and thermal properties. Besides having high sensitivity and operational stability, the fibrous sensor can generate remarkable electrical signals from applied pressure, with an output voltage of 18.1 V, output current of 0.177  $\mu$ A, and power density of 22.9  $\mu$ W cm<sup>-2</sup>. Moreover, such sensors also produce significant electrical performance of up to a few volts under human mechanical stress, thereby allowing for the monitoring of biomechanical movement of the human foot, elbow, and finger. Our study sheds light onto the use of piezoelectric polymers for flexible self-powered sensing electronics and wearable devices.

Received 13th April 2020

Accepted 2nd June 2020

DOI: 10.1039/d0ra03293j

rsc.li/rsc-advances

## Introduction

The development of flexible multifunctional sensors as essential elements in wearable devices and robotic applications, has attracted extensive attention in recent years.<sup>1–3</sup> In particular, the increasing demand for wearable sensors in healthcare, robotics and biomedicine requires these sensors to have higher sensitivity, flexibility, and low-cost. For example, these smart sensors can be utilized to detect the strains or stresses induced by human activity, offering a fast and convenient method to track human motions, such as running and breathing. Although we have witnessed the rapid developments of these sensors recently, fabrication of high accuracy and high-resolution sensing devices, particularly able to make quantitative measurements over a wide range of parameters, remains an elusive, long-term challenge.<sup>4,5</sup> Furthermore, most sensors need to mimic human skin and integrate with other components such as batteries, satisfying the requirements for conformable contact and self-sustainable power source. Therefore, pressure sensors offer new opportunities in human healthcare and activity monitoring.

State-of-the-art sensors mainly rely on the mechanism resulting from the variations of force-related parameter, such as capacitance, resistance, pyroelectric or piezoelectric output and triboelectric output.<sup>6</sup> Notably, some limitations on sensitivity and power consumption are reported in both capacitor and transistor-based sensors. By contrast, piezoelectric polymer has great potential to fulfil the requirements of being mechanically flexible, light weight and operationally stable. Furthermore, it allows for the sustainable operation without an external battery.<sup>7,8</sup> Among them, poly(vinylidene fluoride) (PVDF) has been widely investigated for non-volatile memory, touch display device, and wearable piezoelectric generator owing to its excellent flexibility, piezoelectricity and non-toxicity.<sup>9–11</sup> More importantly, PVDF-based devices can easily realize the self-powered multifunctional sensing due to their piezoelectricity. Its co-polymer, poly(vinylidene fluoride-co-trifluoroethylene) (P(VDF-TrFE)) can easily form pure  $\beta$  phase with the addition of a small amount of trifluoroethylene (TrFE), resulting in better crystallinity and piezoelectricity than PVDF.<sup>12</sup> Particularly, highly aligned P(VDF-TrFE) integrated with 3D electrodes exhibits good flexibility and improved output performances.<sup>13</sup> Besides, with the addition of graphene or BaTiO<sub>3</sub>, output performances of P(VDF-TrFE) fibers were significantly enhanced.<sup>14,15</sup> Novel synthesis technique such as 3D printing can lead to a broad range of strain operation, ranging of up to 300% in BaTiO<sub>3</sub> and P(VDF-TrFE) matrix, which is suitable for industrial manufacturing processes.<sup>16,17</sup> Therefore, piezoelectric polymer such as P(VDF-TrFE) is an attractive candidate for wearable electronics and stretchable energy harvesters in our daily life.

<sup>a</sup>Temasek Laboratories, Nanyang Technological University, Research Techno Plaza, 50 Nanyang Drive, 637553, Singapore. E-mail: mmzhu@ntu.edu.sg; duzehui@ntu.edu.sg

<sup>b</sup>NOVITAS, School of Electrical and Electronic Engineering, Nanyang Technological University, 50 Nanyang Avenue, 639798, Singapore

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ra03293j



Herein, we have constructed a wafer-scale and self-powered pressure sensor based on P(VDF-TrFE) nanofibers by electrospinning technique. The fibers are pure  $\beta$  phase and exhibit excellent mechanical and thermal properties. The resulting sensors demonstrate a strong electrical voltage of around 18.1 V and output current of 0.177  $\mu$ A, corresponding to the power density of 22.9  $\mu$ W cm<sup>-2</sup>. Moreover, we analyzed the pressure sensing mechanisms quantitatively, thereby highlighting the potential for self-powered micro-mechanical sensor, self-balancing soft robots and impact detectors. When it is attached to the human skin, the piezoelectric output signals are observed at the foot, the elbow, and the finger under differential human body motions. This study sheds a deeper insight into self-powered and sensitive impact pressure sensors, paving the way for advanced sensor technologies for applications in artificial intelligence, human-machine interfaces, and healthcare monitoring devices.

## Experimental

### Nanofiber fabrication

P(VDF-TrFE) powders were purchased from Solvay company, with a purity of around 99.99%. In our work, P(VDF-TrFE) nanofibers were synthesized by electrospinning technique. In a typical electrosun procedure, 5 g of P(VDF-TrFE) (75/25 mol%) was dissolved into *N,N*-dimethylformamide (DMF, Sigma-Aldrich, Singapore). Subsequently, this compound was stirred by a magnetic agitator for 6 hours to form a homogeneous solution, which was used as the precursor for the electrospinning. The precursor mixture was pumped into a 15 mL plastic syringe with a metal needle. The electrospinning voltage was 28 kV while the distance between the needle and the cylinder collector was 110 cm. The feed rate of the precursor solution was 2 mL h<sup>-1</sup>, and the rotating speed of the cylinder collector was 1000 revolutions per second.

### Characterizations

The morphology and microstructure of the as-grown fibrous samples were characterized by scanning electron microscopy (FESEM6340, Oxford) and X-ray diffraction (XRD, D8Advance, Bruker). Fourier-Transform Infrared Spectroscopy (FTIR, Thermo Fisher Scientific, USA) was carried out in the range of 400 cm<sup>-1</sup> to 4000 cm<sup>-1</sup>. X-ray Photoelectron Spectroscopy (XPS) was performed to investigate the chemical bonding and composition. In addition, Differential Scanning Calorimetry (DSC) technique were performed to determine the Curie temperature and other thermal decomposition performance. Mechanical properties of the samples are carried out by Instron mechanical tester with 30 kN load force. A customer-designed machine equipped with one resistance box and one computer was used to investigate the generated voltage and output current of the resulting sensor based on the fibrous samples with Cu and Au electrodes (Fig. S1†). In our setup, a portable platform with Au-coated Si wafer (the left platform) as one electrode driven by one motor moves forward and backward, which periodically contacts and releases the P(VDF-TrFE)

nanofibers with copper tape (the right platform) as another electrode. Notably, the load force is associated with the distance between the left sample platform and the right Au/Si platform. The current and voltage outputs were recorded by the computer in tandem with the periodic motion.

## Results and discussion

Electrospinning can be viewed as an electrohydrodynamic process. Under the applied high voltage, the charged liquid droplet generates the jet, which goes through a spherical path and forms the fiber membrane on the rotating collector.<sup>18</sup> Fig. 1a shows the schematic diagram of the electrospinning setup. The machine used in the experiments consists of one plastic syringe with a metal needle, rotating collector and a high power supply (Fig. 1b).<sup>19</sup> It can be seen from Fig. 1c that the size of such nanofibers can reach tens of centimeters and an even larger area which depends on the collector size. More importantly, the excellent fibrous structure forms are observed, evidenced by SEM characterization, as shown in Fig. 1d. Meanwhile, the fiber diameter shows good uniformity and is evaluated to be 200.3 nm, which is consist with the values reported in the literature of up to a few hundred nanometers.<sup>20</sup> In fact, the diameter of electrospinning nanofibers is associated with some experimental factors, such as the precursor concentration, solution viscosity, surface tension, applied voltage and the distance between the needle and the collector. When the concentration of the precursor solution is low, bead formation easily occurs on the fibers, as a result of the Taylor cone instability induced by the imbalance between the viscoelastic and electrical properties of the solution.<sup>21</sup> The diameter ( $d$ ) of the fiber is determined by the equilibrium between the coulombic repulsion and the surface tension,<sup>22</sup> and can be expressed as:  $d = \gamma^{1/3}(Q/I)^{2/3}\omega^{1/2}$ , where  $\gamma$  is the surface tension,  $Q$  is the

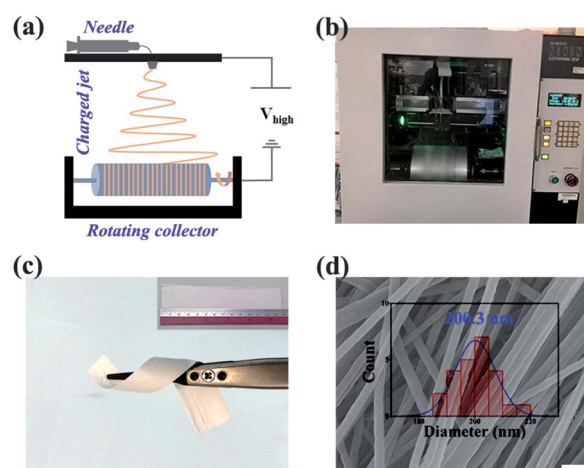


Fig. 1 P(VDF-TrFE) nanofiber fabrication by electrospinning. (a) Schematic illustration and (b) the photograph of the setup for the electrospinning procedure. (c) Photograph of the free-standing fibrous membrane. The inset reveals the large-size sample of up to tens of centimeters. (d) SEM image of the as-synthesized P(VDF-TrFE) nanofibers. The inset is the fiber diameter distribution in the samples. The scale bar denotes 1  $\mu$ m.



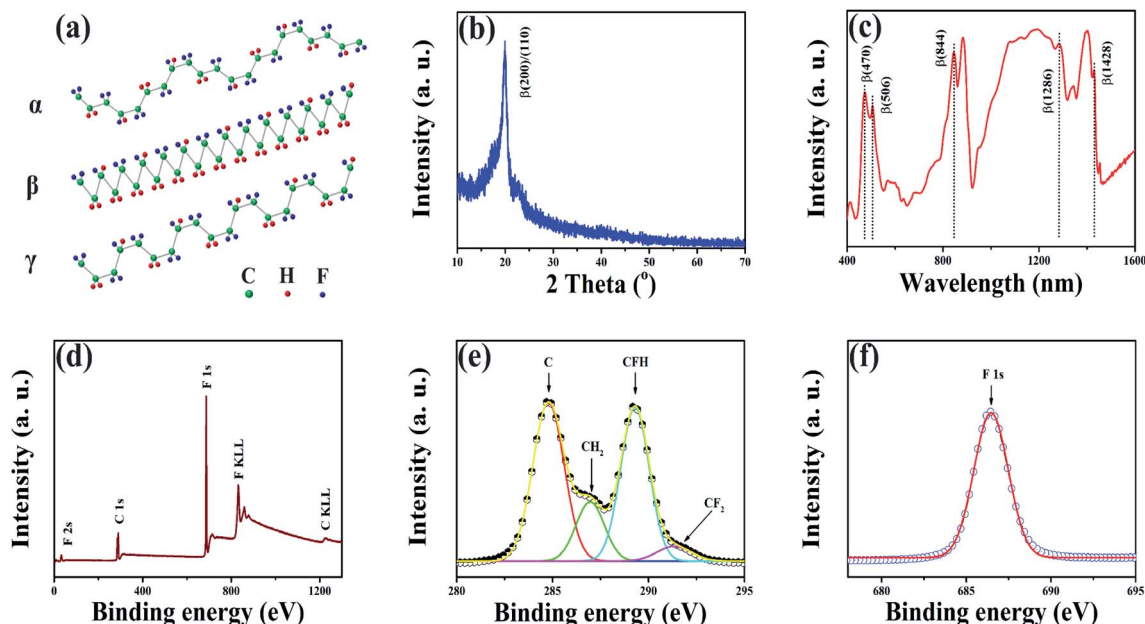


Fig. 2 Crystallinity and phase identification of electrospinning P(VDF-TrFE) nanofibers. (a) Schematic representation of the chemical chain structures for  $\alpha$ ,  $\beta$ , and  $\gamma$  phases of P(VDF-TrFE). (b) XRD pattern, (c) FTIR spectra and (d) XPS full survey scanning spectra of the samples. High resolution XPS spectra of (e) C 1s and (f) F 1s.

volumetric flow rate,  $I$  is the electric current and  $w$  is the initial polymer volume fraction. The distance between the tip and the collector also affect the fiber diameter, owing to the electric field strength variation.<sup>23</sup> Furthermore, the nature of the solvent also influences the diameter because it directly affects the precursor solution surface tension.<sup>24</sup>

Ferroelectric P(VDF-TrFE) copolymer is expected to be more crystalline as compared to PVDF, with the addition of TrFE.<sup>25</sup> As a semi-crystalline polymer, there are five different polymorphs, such as  $\alpha$ ,  $\beta$ ,  $\gamma$ ,  $\delta$ , and  $\epsilon$  phase.<sup>26</sup> Among them,  $\alpha$ ,  $\beta$ , and  $\gamma$  are the most common forms. Notably,  $\delta$  phase is difficult to verify experimentally and formed only by applying a short electrical

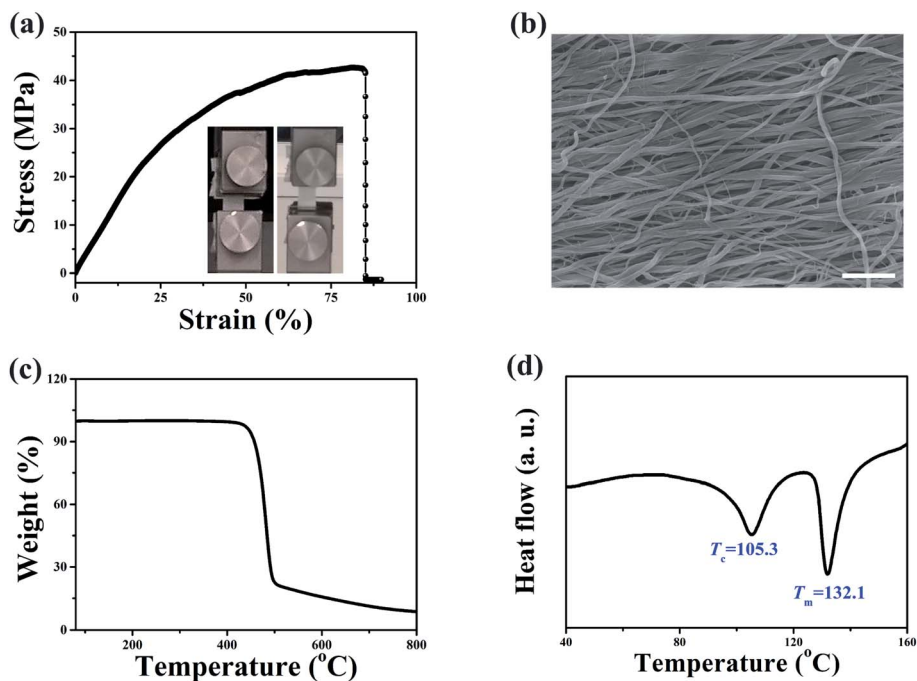


Fig. 3 Mechanical and thermal characterization of the piezoelectric fibers. (a) Stress-strain curves of P(VDF-TrFE) nanofibers. The inset are the photographs of the nanofiber before and after the mechanical testing. (b) SEM image of the samples after tensile testing. The scale bar is 1  $\mu\text{m}$ . (c) Thermogravimetric analysis and (d) DSC thermograms of P(VDF-TrFE) nanofibers.

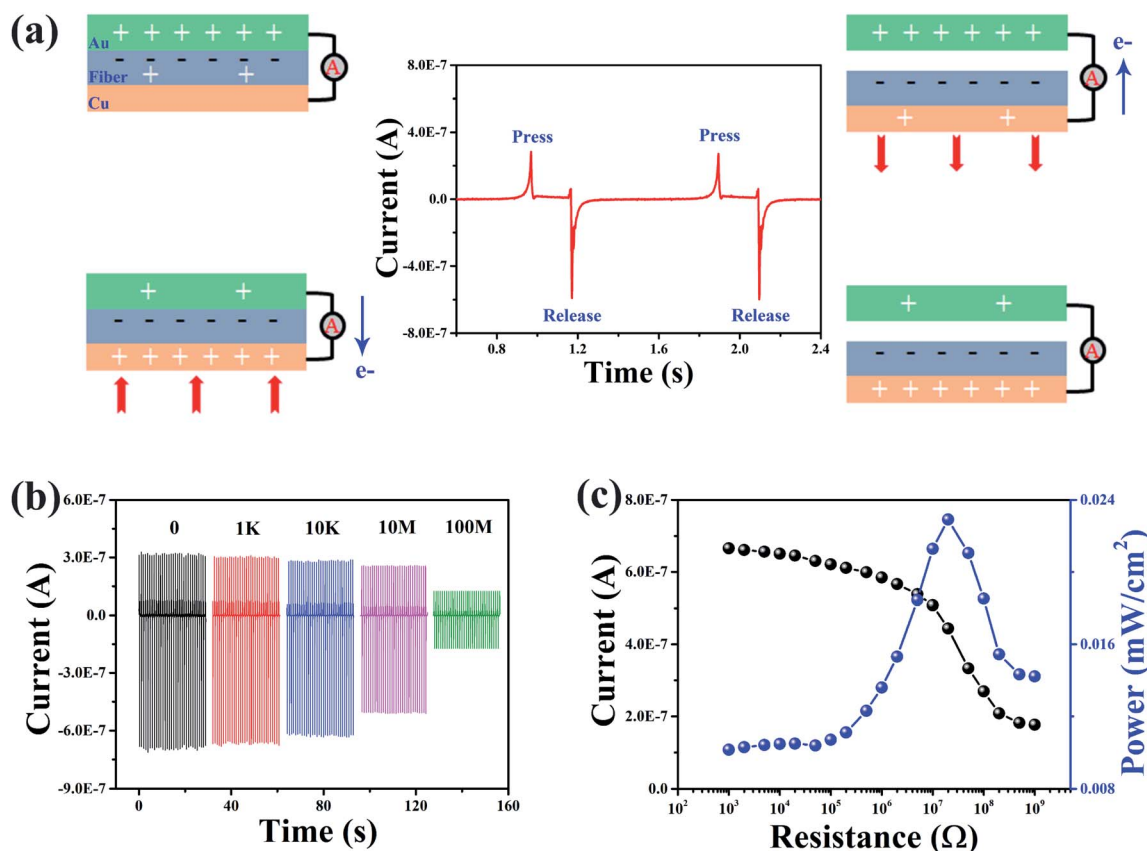


Fig. 4 (a) Schematic of the charge formation and transport in the device during the pressing and releasing process. The inset is the short-circuit current under the press and release actions. (b) Measured output current under different load resistances. (c) Output current and power density of the device as a function of the loaded resistance.

pulse.<sup>27</sup> Although  $\alpha$  phase is the most stable phase thermodynamically,  $\beta$ -phase exhibits better ferroelectric properties and spontaneous polarization.<sup>12,28</sup> Fig. 2a reveals the chemical chain structures of ferroelectric P(VDF-TrFE) polymers with different phases. The crystalline characteristics of P(VDF-TrFE) nanofibers are further studied by XRD, FTIR and XPS. Fig. 2b exhibits the single XRD peak at  $19.9^\circ$ , which is corresponding to the (110)/(200) of  $\beta$  phase.<sup>28</sup> Comparatively, any peaks at  $17.8^\circ$  (100),  $18.5^\circ$  (020),  $26.6^\circ$  (021) for  $\alpha$  phase or the ones at  $18.5^\circ$  (020) and  $20.4^\circ$  (110) for  $\gamma$  phase are not observed, indicating the  $\beta$  phase is the dominant polymorph in our nanofibers.<sup>29,30</sup> Besides, FTIR analysis shows the typical  $\beta$  phase bands at 506, 844, 1286 and  $1428\text{ cm}^{-1}$  (Fig. 2c).<sup>31</sup> By contrast, the  $\alpha$  phase bands ( $532, 612, 765, 796, 870, 970\text{ cm}^{-1}$ ) and  $\gamma$  phase bands ( $1170, 1220\text{ cm}^{-1}$ ) are not present.<sup>31,32</sup> In addition, XPS result also indicates the presence of C and F (Fig. 2d). Moreover, both C 1s and F 1s spectra of the fibers are shown in Fig. 2e and f, respectively. Four prominent C states are recognized (*e.g.*, 284.8 eV for surface contaminants, 286.9 eV for hydrocarbon  $\text{CH}_2$ , 286.3 eV for C-F-H and 291.3 eV for  $\text{CF}_2$ ), which are in agreement with the values of P(VDF-TrFE) reported.<sup>33,34</sup>

Mechanical robustness of piezoelectric P(VDF-TrFE) copolymers is one of the key factors in multifunctional applications. All testing samples are regular squares with dimensions of  $10 \times 10 \times 0.04\text{ cm}^3$ . Fig. 3a depicts the stress-strain curve of the

samples. The tensile strength and Young's modulus of the nanofibers are 42.8 MPa and 157.3 MPa, respectively, which are similar to those of P(VDF-TrFE) nanofibers by electrospinning and significantly larger than the value of P(VDF-TrFE) thin films.<sup>35,36</sup> Fig. 3b displays the morphology of the sample after tensile testing. Apparently, the nanofiber becomes flatten extensively, indicating the excellent elasto-plastic behavior. As shown in Fig. 3c, the initial thermal decomposition temperature of the fibrous samples is around  $420^\circ\text{C}$ . Fig. 3d shows the Curie temperature ( $T_c$ ,  $105.3^\circ\text{C}$ ) and melting transition point ( $T_m$ ,  $132.1^\circ\text{C}$ ) of P(VDF-TrFE) nanofibers. It is noted that  $T_c$  is characteristic of the ferroelectric behaviors, depending on the morphology of piezoelectric polymer (size, amount, molecular weight, cooperativity *etc.*).<sup>26,37</sup> Besides, it is believed that the ratio of VDF to TrFE significantly influences the  $T_c$  and  $T_m$ .<sup>38</sup> As the amount of VDF increases from 80 to 94 mol%,  $T_c$  increases from  $90.9$  to  $124.7^\circ\text{C}$  whereas  $T_m$  slightly increases by around  $6^\circ\text{C}$  ( $140.7$  to  $146.02^\circ\text{C}$ ).

Periodical press and release actions were carried out on the device to investigate the output electrical performances. As shown in Fig. 4a, P(VDF-TrFE) nanofibers were in contact with the bottom electrode copper and the top electrode gold. When the device is pressed initially, electrons are injected from the top electrode to the P(VDF-TrFE) surface, followed by the charge transferring to the bottom electrode during the release process.





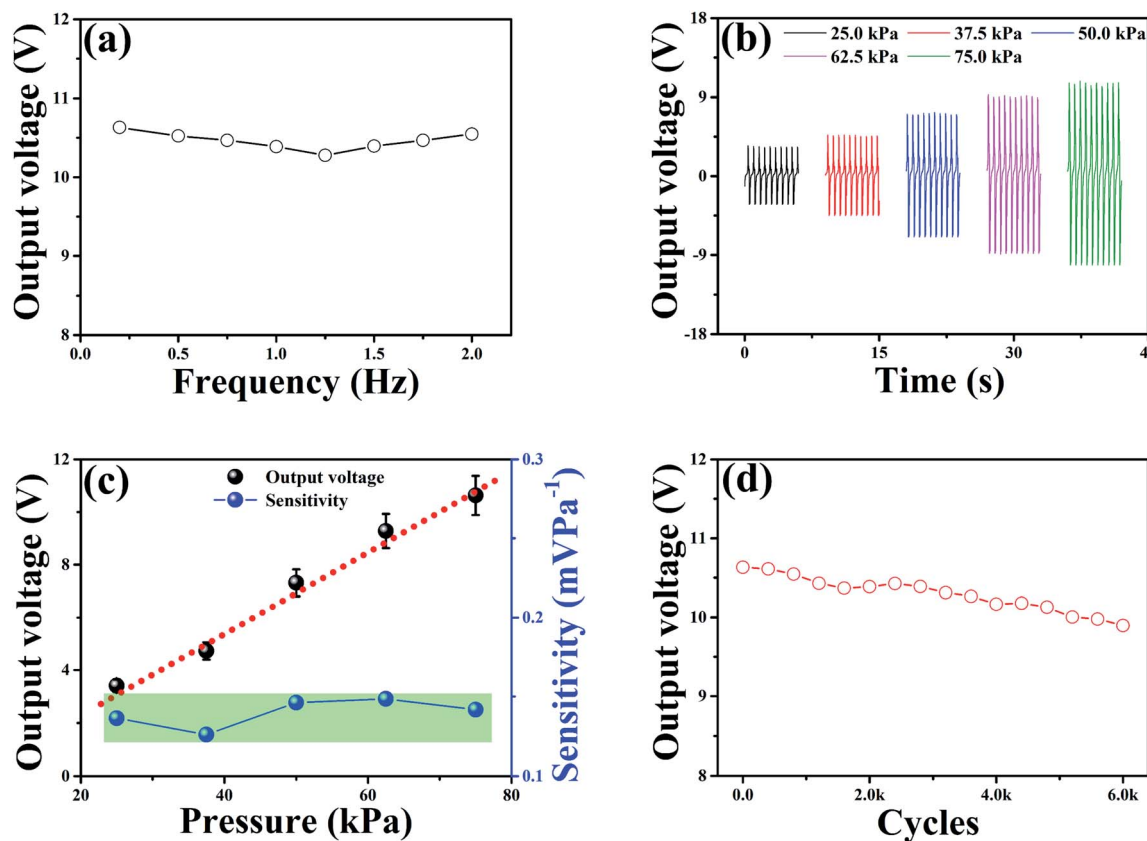


Fig. 5 Pressure sensor performances of the device. (a) Output voltage as a function of frequency under a pressure of 75 kPa. (b) Output voltage under different loading forces. (c) Relationship between the applied pressure and the output voltage, as well as the corresponding sensitivity. (d) Stability testing of the output voltage in our sensing device.

Apparently, the voltage potential between two electrodes establishes, allowing the electrons to flow through the top electrode to the bottom electrode in order to remain the electrostatic equilibrium.<sup>39</sup> Once the device is pressed again, the redistributed charges are created to drive the electrons in the opposite direction. Notably, spontaneous induction of polarity of the fibers occur easily at high electric field during the electrospinning process, although no polarity of the fibers are induced before the measurement.<sup>40</sup> Fig. 4b exhibits the output current variation with external load resistances ranging from 1 k $\Omega$  to 1 G $\Omega$ . The corresponding output current and power density are summarized in Fig. 4c.<sup>41</sup> It seems clear that the output current continues to decrease with the external load resistance. By contrast, the instantaneous output power density slightly increases at the beginning and subsequently reaches the maximum value at the load resistance of 10 M $\Omega$ , resulting in a maximum output power of 22.9  $\mu\text{W cm}^{-2}$ . Similar phenomenon of a first increase and then decrease in power density is also observed in ZnO-modified P(VDF-TrFE) thin films and nanofibers.<sup>42,43</sup>

Pressure sensors can quickly detect small pressure variations, indicating the immense potential in soft robotics, human-machine interfaces and artificial intelligence, as well as healthcare monitoring devices. Herein, we investigate the sensing mechanism of the fiber-based device. A dynamic force

measurement was carried out under a pressure of 75 kPa at the frequency range of 0.2 to 2 Hz (Fig. 5a). Notably, the output voltage remains relatively constant with the increase of the frequency. Interestingly, as the external load pressure increases by a larger margin, the output voltage of the device exhibits a similar increasing trend (Fig. 5b). Here, the sensitivity is defined as the ratio of the voltage and the pressure ( $V/P$ ). All output voltage and sensitivity of our device are calculated and summarized in Fig. 5c. In general, the output voltage is associated with the piezoelectric charge coefficient ( $d_{33}$ ) and the applied pressure ( $P$ ):<sup>44</sup>

$$V = (hd_{33}P)/(\epsilon_0\epsilon_r) \quad (1)$$

where  $h$  is the thickness,  $d_{33}$  is the piezoelectric constant of the material,  $\epsilon_0$  is the permittivity of free space, and  $\epsilon_r$  is the relative dielectric constant of the piezoelectric materials. At a pressure of 25.0 kPa, the generated voltage of our device is calculated to be around 3.4 V, which corresponds to a sensitivity of 141.7 mV kPa<sup>-1</sup>. In addition, the measured output voltages of the sensor with various load pressures of 25.0, 37.5, 50.0, 62.5 and 75.0 kPa are shown in Fig. 5c. The experimental results clearly reveal that the output voltage is linearly related with the pressure and agrees well with the eqn (1). Moreover, the maximum voltage of the sensing device is around 10.6 V at a pressure of 75 kPa. All



Table 1 Summary of the characteristics of P(VDF-TrFE)-based nanofiber in this work and other reported studies for sensor devices

Materials	Method	Form	Output voltage (V)	Sensitivity (mV Pa <sup>-1</sup> )	References
MWCNT/P(VDF-TrFE)	Electrospinning	Fiber	6.2	0.20	45
Graphene/P(VDF-TrFE)	Spincoating	Film	<1	—	14
Graphene/P(VDF-TrFE)	Electrospinning	Fiber	1.5	0.072	46
BaTiO <sub>3</sub> /P(VDF-TrFE)	3D printing	Film	8.0	—	16
BaTiO <sub>3</sub> /P(VDF-TrFE)	Electrospinning	Fiber	6.0	—	15
BN/P(VDF-TrFE)	Spincoating	Film	22	0.055	47
Ag/P(VDF-TrFE)	Spincoating	Film	17	0.1	48
P(VDF-TrFE)	Printing	Film	1	0.066	49
P(VDF-TrFE)	Electrospinning	Fiber	4.2	0.41	31
P(VDF-TrFE)	Electrospinning	Fiber	10.6	0.14	This work

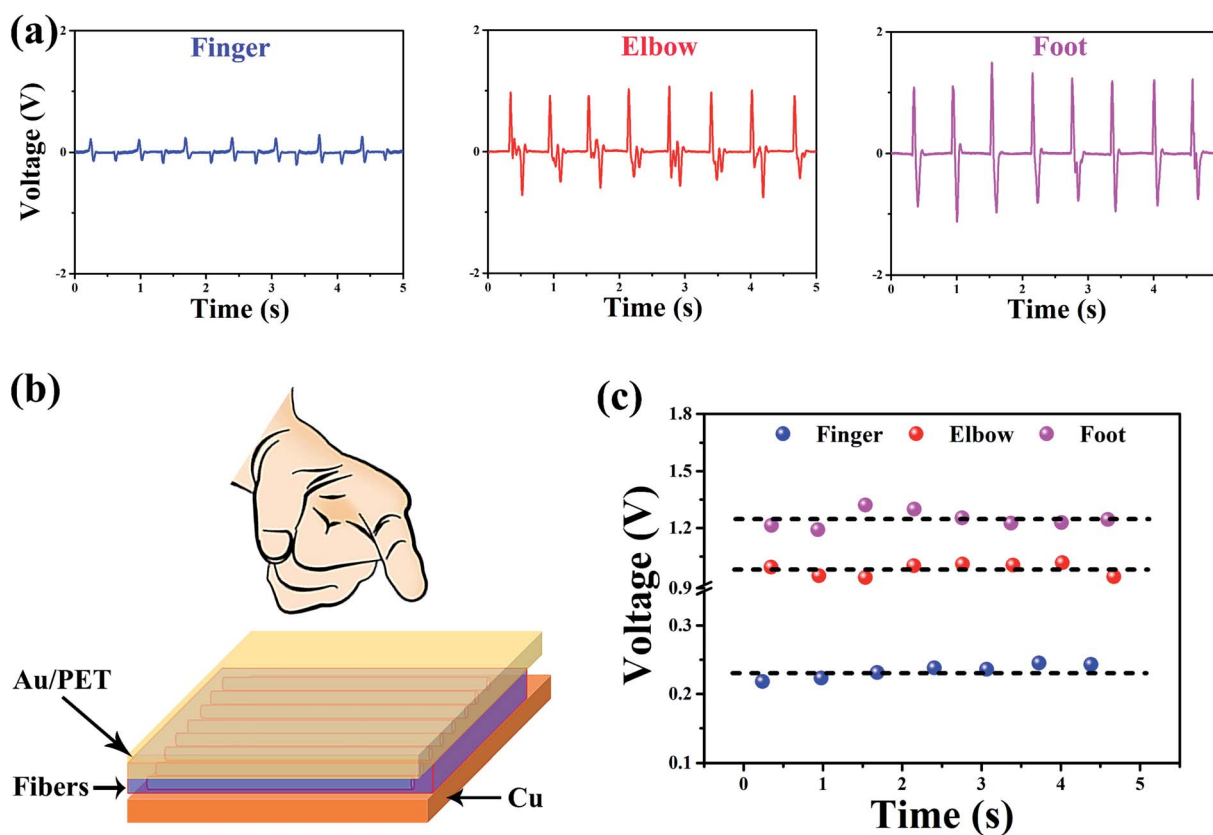


Fig. 6 Applications of self-powered human activity monitoring. (a) Output voltage signals of the sensor device at human body parts of finger, elbow, and foot. (b) Schematic illustration of human movement with the device. (c) The statistic results for output voltage induced by human activity movement, indicating the operation stability of our device.

the characteristics of our P(VDF-TrFE) fiber sensor in comparison with other reported studies are summarized in Table 1. Fig. 5d displays the stability response of the sensor at a pressure of 75 kPa. The presence of only 5% attenuation for the generated voltage after 5000 cycles of repeated measurement indicates the excellent reliability of our sensor devices and holds great promise for further commercialization.

Owing to excellent mechanical flexibility and high sensitivity of P(VDF-TrFE)-based sensor above, it has tremendous potential in tracking human body motion. Therefore, we attach the devices on different parts of the human body, such as foot,

elbow and finger to study its practical performances. Tensile and compressive stresses periodically occur in tandem with human body movements, thereby driving the device to generate cyclic electrical response, as shown in Fig. 6a. Schematic illustrations of human motions with the device attached to the finger can be found in Fig. 6b. Apparently, the foot exhibits a largest output voltage as compared to the elbow and the finger, with a peak open-circuit voltage of  $\sim 1.3$  V. More importantly, our device exhibits a fast response and recovery time in ten milliseconds (Fig. S2†). The output performance in the finger also demonstrates a similar trend, however, the value



of its open-circuit voltage is only around 0.23 V, which is far smaller than those of the elbow case (0.98 V) and the foot case (1.3 V). Fig. 6c displays the results of the peak open-circuit voltages in a short time, steadily exhibiting the values of around 1.3 V for the foot, 0.98 V for the elbow and 0.23 V for the finger. Our device does not only offer an excellent sensitivity in the generated electrical signal, but also reveals a quantitative difference on various body parts, which can pave the ways for multifunctional applications in soft robots and wearable electronic devices.

## Conclusions

In conclusion, we have presented the functionality of human motion-inspired sensors based on electrospinning P(VDF-TrFE) nanofibers. These fibrous membranes exhibit a pure  $\beta$ -phase and excellent mechanical flexibility, as well as a uniform fibrous structure. The device can directly convert the external pressure to the electrical signals, allowing the instantaneous detection of stress stimuli. Besides having high sensitivity on pressure sensing, such sensor offers fast response speed, excellent operation stability and self-powered capability. The piezoelectric output signals, when the sensors are attached to the human body at different locations, were stabilized at 1.3 V (foot), 0.98 V (elbow) and 0.23 V (finger) under differential human movements. Piezoelectric polymer nanofibers can achieve highly sensitive energy harvesting capacity for human body motions under various stress, paving the way for self-powered e-skins and multifunctional wearable micro-/nano-electronic devices.

## Conflicts of interest

There are no conflicts to declare.

## Notes and references

- J. Xiong, P. Cui, X. Chen, J. Wang, K. Parida, M.-F. Lin and P. S. Lee, *Nat. Commun.*, 2018, **9**, 4280.
- E. Roh, B. Hwang, D. Kim, B. Kim and N. Lee, *ACS Nano*, 2015, **9**, 6252–6261.
- Y. C. Huang, Y. Liu, C. Ma, H. C. Cheng, Q. He, H. Wu, C. Wang, C. Y. Lin, Y. Huang and X. Duan, *Nat. Electron.*, 2020, **3**, 59–69.
- H. Zou, Y. Zhang, L. Guo, P. Wang, X. He, G. Dai, H. Zheng, C. Chen, A. C. Wang, C. Xu and Z. L. Wang, *Nat. Commun.*, 2019, **10**, 1427.
- M. Peng, Z. Li, C. Liu, Q. Zheng, X. Shi, M. Song, Y. Zhang, S. Du, J. Zhai and Z. L. Wang, *ACS Nano*, 2015, **9**, 3143–3150.
- W. Deng, T. Yang, L. Jin, C. Yan, H. Huang, X. Chu, Z. Wang, D. Xiong, G. Tian, Y. Gao, H. Zhang and W. Yang, *Nano Energy*, 2019, **55**, 516–525.
- E. J. Ko, S. J. Jeon, Y. W. Han, S. Y. Jeong, C. Y. Kang, T. H. Sung, K. W. Seong and D. K. Moon, *Nano Energy*, 2019, **58**, 11–22.
- Y. Xin, C. Guo, X. Qi, H. Tian, X. Li, Q. Dai, S. Wang and C. Wang, *Ferroelectrics*, 2016, **500**, 291–300.
- J. Y. Son, S. Ryu, Y. C. Park, Y. T. Lim, Y. S. Shin, Y. H. Shin and H. M. Jang, *ACS Nano*, 2010, **4**, 7315–7320.
- S. Reis, V. Correia, M. Martins, G. Barbosa, R. M. Sousa, G. Minas, S. Lanceros-Mendez and J. G. Rocha, in *2010 IEEE International Symposium on Industrial Electronics*, IEEE, 2010, pp. 516–520.
- F. Chen, Y. Wu, Z. Ding, X. Xia, S. Li, H. Zheng, C. Diao, G. Yue and Y. Zi, *Nano Energy*, 2019, **56**, 241–251.
- G. X. Ni, Y. Zheng, S. Bae, C. Y. Tan, O. Kahya, J. Wu, B. H. Hong, K. Yao and B. Özyilmaz, *ACS Nano*, 2012, **6**, 3935–3942.
- L. Zhang, J. Gui, Z. Wu, R. Li, Y. Wang, Z. Gong, X. Zhao, C. Sun and S. Guo, *Nano Energy*, 2019, **65**, 103924.
- S. Kim, Y. Dong, M. M. Hossain, S. Gorman, I. Towfeeq, D. Gajula, A. Childress, A. M. Rao and G. Koley, *ACS Appl. Mater. Interfaces*, 2019, **11**, 16006–16017.
- X. Guan, B. Xu and J. Gong, *Nano Energy*, 2020, **70**, 104516.
- X. Zhou, K. Parida, O. Halevi, Y. Liu, J. Xiong, S. Magdassi and P. S. Lee, *Nano Energy*, 2020, **72**, 104676.
- B. Stadlober, M. Zirkel and M. Irimia-Vladu, *Chem. Soc. Rev.*, 2019, **48**, 1787–1825.
- J. Xue, T. Wu, Y. Dai and Y. Xia, *Chem. Rev.*, 2019, **119**, 5298–5415.
- Z.-D. Liu, Z.-Y. Yin, Z.-H. Du, Y. Yang, M.-M. Zhu, L.-H. Xie and W. Huang, *Nanoscale*, 2014, **6**, 5110.
- G. Ico, A. Showalter, W. Bosze, S. C. Gott, B. S. Kim, M. P. Rao, N. V. Myung and J. Nam, *J. Mater. Chem. A*, 2016, **4**, 2293–2304.
- H. Fong, I. Chun and D. Reneker, *Polymer*, 1999, **40**, 4585–4592.
- R. Stepanyan, A. V. Subbotin, L. Cuperus, P. Boonen, M. Dorsch, F. Oosterlinck and M. J. H. Bulters, *Polymer*, 2016, **97**, 428–439.
- M. M. Hohman, M. Shin, G. Rutledge and M. P. Brenner, *Phys. Fluids*, 2001, **13**, 2221–2236.
- A. Haider, S. Haider and I. K. Kang, *Arabian J. Chem.*, 2018, **11**, 1165–1188.
- B. Chu, *Science*, 2006, **313**, 334–336.
- Prateek, V. K. Thakur and R. K. Gupta, *Chem. Rev.*, 2016, **116**, 4260–4317.
- M. Li, H. J. Wondergem, M. J. Spijkman, K. Asadi, I. Katsouras, P. W. M. Blom and D. M. De Leeuw, *Nat. Mater.*, 2013, **12**, 433–438.
- M. Zhu, J. Wu, Z. Du, R. Y. Tay, H. Li, B. Özyilmaz and E. H. T. Teo, *Nanoscale*, 2015, **7**, 14730–14737.
- Z. Y. Jiang, X. C. Zheng and G. P. Zheng, *RSC Adv.*, 2015, **5**, 61946–61954.
- W. Xia, Z. Wang, J. Xing, C. Cao and Z. Xu, *IEEE Trans. Ultrason. Ferroelectr. Freq. Control*, 2016, **63**, 1674–1680.
- L. Persano, C. Dagdeviren, Y. Su, Y. Zhang, S. Girardo, D. Pisignano, Y. Huang and J. A. Rogers, *Nat. Commun.*, 2013, **4**, 1633.
- A. N. Arshad, M. H. M. Wahid, M. Rusop, W. H. A. Majid, R. H. Y. Subban and M. D. Rozana, *J. Nanomater.*, 2019, **2019**, 1–12.
- Z. Zhang, Y. Zhao and M. Zhu, *Appl. Phys. Lett.*, 2006, **88**, 033101.



- 34 Y. Y. Choi, J. Hong, D. S. Leem, M. Park, H. Song, T. H. Sung and K. No, *J. Mater. Chem.*, 2011, **21**, 5057–5061.
- 35 Y. Yao, G. Hou, N. Li, T. Zhou, L. Liu, Y. Liu and J. Leng, *J. Appl. Polym. Sci.*, 2017, **134**, 1–11.
- 36 S. Park, Y. Kwon, M. Sung, B. S. Lee, J. Bae and W. R. Yu, *Mater. Des.*, 2019, **179**, 107889.
- 37 C. Wan and C. R. Bowen, *J. Mater. Chem. A*, 2017, **5**, 3091–3128.
- 38 W. M. Xia, Y. J. Gu, C. Y. You, C. J. Cao, Z. Xu and Z. C. Zhang, *RSC Adv.*, 2015, **5**, 107557–107565.
- 39 Z. L. Wang, J. Chen and L. Lin, *Energy Environ. Sci.*, 2015, **8**, 2250–2282.
- 40 B. Ameduri, *Chem. Rev.*, 2009, **109**, 6632–6686.
- 41 M. M. Abolhasani, M. Naebe, K. Shirvanimoghaddam, H. Fashandi, H. Khayyam, M. Joordens, A. Pipertzis, S. Anwar, R. Berger, G. Floudas, J. Michels and K. Asadi, *Nano Energy*, 2019, **62**, 594–600.
- 42 J. Li, C. Zhao, K. Xia, X. Liu, D. Li and J. Han, *Appl. Surf. Sci.*, 2019, **463**, 626–634.
- 43 H. Yuan, T. Lei, Y. Qin and R. Yang, *Nano Energy*, 2019, **59**, 84–90.
- 44 R. A. Surmenev, T. Orlova, R. V. Chernozem, A. A. Ivanova, A. Bartasyte, S. Mathur and M. A. Surmeneva, *Nano Energy*, 2019, **62**, 475–506.
- 45 A. Wang, M. Hu, L. Zhou and X. Qiang, *Nanomaterials*, 2018, **8**, 1021.
- 46 P. Li, L. Zhao, Z. Jiang, M. Yu, Z. Li and X. Li, *Macromol. Mater. Eng.*, 2019, **304**, 1–8.
- 47 S. Ye, C. Cheng, X. Chen, X. Chen, J. Shao, J. Zhang, H. Hu, H. Tian, X. Li, L. Ma and W. Jia, *Nano Energy*, 2019, **60**, 701–714.
- 48 C. K. Jeong, D. Y. Hyeon, G. T. Hwang, G. J. Lee, M. K. Lee, J. J. Park and K. Il Park, *J. Mater. Chem. A*, 2019, **7**, 25481–25489.
- 49 T. Sekine, R. Sugano, T. Tashiro, J. Sato, Y. Takeda, H. Matsui, D. Kumaki, F. D. Dos Santos, A. Miyabo and S. Tokito, *Sci. Rep.*, 2018, **8**, 1–10.

