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Introduction

Heterogeneous semiconductor nanowire array for sensitive broadband photodetector by crack photolithography-based micro-/nanofluidic platforms⁺

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The *in situ* growth of nanowires (NWs) into nano-/microelectromechanical systems (NEMS/MEMS) by solution processing is attractive for its relative simplicity and economic value. We present innovative, versatile microfabrication that produces multiple, heterogeneous semiconductor NWs. A crack photolithography-based micro-/nanofluidic platform has been developed. This platform offers the *in situ* solution growth of NWs while enabling the control of quantity, dimensions, orientation, alignment, position, and material. The generation of grain boundary (GB)-rich CH₃NH₃Pbl₃ NWs using the micro-/nanofluidic device is exemplary. High-quality single-crystal CH₃NH₃Pbl₃ NWs were derived by injection of a CH₃NH₃Pbl₃ solution. We produced a parallel NW array of CH₃NH₃Pbl₃ NW for visible light detection and ZnO NW for ultraviolet detection, thereby demonstrating an unprecedented broadband composite photodetector. The micro-/nanofluidic fabrication platform enables the production of multiple, heterogeneous semiconductor NW arrays on one substrate, offering the potential to elicit synergistic performance and functional enhancements from various NWs.

systems (NEMS/MEMS), which are integrated miniature devices that combine electrical and mechanical components at the nanoscale,¹² is a significant process. The complex application of NWs in NEMS/MEMS requires a suitable fabrication process or equally advanced manipulation methods.

Solution processing is a particularly desirable micro-/ nanofabrication technology because of the relative simplicity and potential economy it offers.13 Solution growth of NWs usually has a high yield, but it often produces unwanted products of other morphologies.3 Nanowire manipulation is also necessary in device preparation. In situ solution processing is mainly used in film deposition processes¹⁴ rather than NWs fabrication. NW fabrication by solution processes usually requires a template or pattern that are confined to special substrates or need further transferring process.^{1,15} Onedimensional nanomaterials like carbon nanotubes can be selfassembled on prepatterned substrates with chemical functionalization.¹⁶ In situ growth and the control of semiconductor NWs are still difficult at low processing temperatures. The more considerable challenge is to integrate NWs comprised of different semiconductor materials into a single NEMS/MEMS while keeping processing energy consumption low in conventional micro-/nanofabrication. Such nanosystems can provide a synergistic performance and functional enhancements with multiple, heterogeneous NWs on one substrate.

NW-based photodetector optoelectronic devices have been extensively studied, but the photo response remains limited to

Semiconductor nanowires (NWs) are widely used for elec-

tronics,¹ bio/chemical sensors,² lasers,³ and photodetectors⁴⁻⁷ for their uniquely high surface-to-volume ratio that enables

high sensitivity, rapid response time, and low power

consumption.8 However, their fabrication methods typically rely

on traditional micro-/nanofabrication equipment and facilities

so precisely controlling the number, dimensions, orientation,

alignment, and position of NWs on the same substrate9 remains

a challenge. For example, chemical vapor deposition (CVD) has

the general advantage of precise surface, morphology, structure,

and composition control of a product, thus providing an alter-

native synthesis route toward high quality single-crystalline semiconductor nanowires.^{10,11} However it is difficult to directly

use the nanowires prepared by CVD methods in the devices

based on a single nanowire without manipulation and transfer. The integration of NWs into the nano-/microelectromechanical

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either ultraviolet (UV) or visible light spectra.^{17,18} Some materials, such as perovskites, respond to both ultraviolet and visible light, but they are not very responsive in both areas at the same time.⁶ This limitation can easily be overcome with a photodetection NW array design that uses multiple materials with high responsivities in different wavelength ranges. Such a heterogeneous semiconductor NW based broadband photodetector based on *in situ* synthesis has not yet been demonstrated and reported, mainly because there are no available micro- and/or nanofabrication methods to controllably integrate such a heterogeneous semiconductor NW array in number, orientation, and material in the desired locations in a photodetector.

Here, we show an innovative, versatile microfabrication method that can produce multiple heterogeneous NWs on the same substrate. We apply the crack photolithography¹⁹ fabrication method to produce a parallel NW array of $CH_3NH_3PbI_3$ NWs, a well-known organo-lead halide perovskites,^{20,21} and ZnO NW, thereby creating a broadband photodetector with relatively high sensitivity and responsivity. The photodetector combines the high UV response and detection ability from ZnO NWs and the high visible light response and detection ability from CH_3 -NH₃PbI₃ NWs.

Results and discussion

In Fig. 1a, a schematic of the micro-/nanofluidic device containing a crack nanochannel for NW fabrication is presented. We employ crack-assisted photolithography to prepare a micro-/ nanofluidic device. It facilitates the production of PDMS-based micro-/nanofluidic channel networks using standard photoand soft lithography, as established in our previous work (see photograph in Fig. S1†).¹⁹ The scanning electron microscopy (SEM) image of a typical micro-/nanofluidic device with one central nanochannel is shown in Fig. 1b. The enlarged SEM image (Fig. 1b) and atomic force microscopy (AFM) image (Fig. 1c) indicate that the crack nanochannel is ~2.5 µm wide and 350 nm deep. Two representative methods to fabricate a CH₃-NH₃PbI₃ NWs are performed with the assistance of our micro-/ nanofluidic device.

A series of fabrication processes of CH3NH3PbI3 NWs, called the two-step and open-boundary vapor deposition method (see Experimental section) are illustrated in Fig. 1d. A PbI₂ NW was synthesized by placing drops of a PbI2 solution into the micro-/ nanofluidic device, followed by thermal annealing. The PbI₂ was converted to CH₃NH₃PbI₃, by exposing the PbI₂ NW to CH₃NH₃I vapors in an N₂ container. By this method, the CH₃NH₃PbI₃ NW inherited the grain-rich structure of the PbI₂ NW. An SEM image and a corresponding enlarged image of a PbI2 NW generated after the evaporation of dimethylformamide (DMF), with many small grains are shown in Fig. 1e. The chemical vapor transport reaction transmuted the PbI2 grains into CH3NH3PbI3 grains (Fig. 1f). The grain boundary (GB)-rich structure was also inherited from the PbI2 NW. AFM imaging was conducted to evaluate the dimensions of the CH₃NH₃PbI₃ NWs. As shown in Fig. 1g, the width and height of the CH₃NH₃PbI₃ NW were



Fig. 1 PDMS device with micro-/nanofluidic channel network and (Method-I) nanofluidic $CH_3NH_3PbI_3$ NWs. (a) Schematic of the crackingassisted micro-/nanofluidic device attached to a SiO₂/Si substrate. (b) SEM image of nanowire fabrication from utilizing the nanofluidic channel; enlarged image of the crack channel. (c) AFM image of the crack nanochannel. (d) Two-step open-boundary vapor deposition method (Method-I) fabrication of $CH_3NH_3PbI_3$ NWs. (e) SEM images of the single PbI₂ NW made with the PDMS crack device; close-up view of the PbI₂ NW. (f) $CH_3NH_3PbI_3$ NW produced by further vapor deposition and annealing. (g) AFM image of the GB-rich $CH_3NH_3PbI_3$ NW. (h) TEM image of the GBrich $CH_3NH_3PbI_3$ NW. (i) XRD patterns of PbI₂ NW in (e) and the $CH_3NH_3PbI_3$ NW formed by further vapor deposition and annealing in (f).

approximately 1.5 μ m and 300 nm, respectively (Fig. S2a†). From the transmission electron microscopy (TEM) image in Fig. 1h, it is evident that the rough surface of the CH₃NH₃PbI₃ NW contained many grain boundaries. The purity of phases of the PbI₂ NW before and after the chemical vapor reaction were characterized by X-ray diffraction (XRD). The resulting patterns display the typical peaks of the PbI₂ and CH₃NH₃PbI₃ structures, without impurity peaks (Fig. 1i). This indicates that PbI₂ was completely converted into CH₃NH₃PbI₃.^{22,23}

Our other fabrication process of CH₃NH₃PbI₃ NW using the same micro-/nanofluidic device, called the one-step and closedboundary crystallization method (see Experimental section)²⁴ is shown in Fig. 2. For this method, a CH₃NH₃PbI₃ solution was injected into the micro-/nanofluidic device, and the CH₃NH₃-PbI₃ NW was generated in the crack nanochannel after the evaporation of the solvent in the same N2-flooded container (Fig. 2b). The crack nanochannel appears to restrict the domain size of the CH₃NH₃PbI₃ NW. Bulk CH₃NH₃PbI₃ was also generated in the microchannels on one side of the crack nanochannel. The number of NWs can be controlled by the number of nanochannel in micro-/nanofluidic device. We replaced the SiO₂/Si wafer substrate with a glass slide, and we monitored the real-time crystal growth process of four NWs in one device. The CH3NH3PbI3·DMF NWs crystallization occurred from the end of the nanochannel, where the lower microchannel and the nanochannel met. The NW passed through the nanochannel and continued to grow in the microchannel. The DMF in the microchannels continued to

evaporate, leading to a relatively high concentration of the mixture solution within the microchannels. As a result, crystallization can continue to take place, resulting in bundles of NWs (Fig. 2c and Movie S1†). After annealing, CH₃NH₃PbI₃· DMF NWs will be transformed into CH₃NH₃PbI₃ NWs.

From the enlarged images in Fig. 2d, it is evident that the asprepared $CH_3NH_3PbI_3$ NW had a rather smooth surface. In the AFM image, it can be seen that the as-prepared $CH_3NH_3PbI_3$ NW was arched with a width and height of 2.5 µm and 350 nm, respectively (Fig. 2e and S2b†). The XRD pattern displays the typical peaks of the perovskite structure of $CH_3NH_3PbI_3$ without impurity peaks (Fig. 2f). In Fig. 2g, a typical TEM image of the $CH_3NH_3PbI_3$ NW is shown, with the single-crystal characteristic demonstrated by the selected-area electron diffraction pattern from different positions for the same NW (see Fig. S3†).²⁵

As shown in Fig. 3a, the dark-current of the GB-rich NW shows an obvious hysteresis, whereas no dark-current hysteresis could be observed for the single-crystal NW under the same test conditions (Fig. 3b) although theoretical work predicted a huge Schottky defect density of 0.4% at room temperature.²⁶ But inconspicuous dark-current hysteresis for single-crystal NW suggests that the main reason for this phenomenon can be attributed to faster ion migration at the GBs, which has been reported in studies of both nanoscopic and macroscopic measurements.²⁷

We further investigated ion migration along long NWs with different morphologies (*i.e.*, GB-rich polycrystalline NWs in Fig. 3c, single-crystalline NWs in Fig. 3d and schematically in



Fig. 2 Nanofluidic fabrication process of $CH_3NH_3PbI_3$ NWs (Method-II). (a) Schematic of fabrication of $CH_3NH_3PbI_3$ NW by the one-step closedboundary crystallization method (Method-II). (b) Illustration depicting the growth of a $CH_3NH_3PbI_3 \cdot DMF$ NW. (c) Real-time growth monitoring of a $CH_3NH_3PbI_3 \cdot DMF$ NW from nano- to microchannel network, filled at the upper side. (d) SEM images of the perovskite NW crack channelrestricted domain crystallization method and close-up view of the $CH_3NH_3PbI_3$ NW. (e) AFM image of the single-crystal $CH_3NH_3PbI_3$ NW fabricated by Method-II and NW after air storage in darkness for 100 days. (g) TEM image of the single-crystal $CH_3NH_3PbI_3$ NW.

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Fig. 3 Characterization of $CH_3NH_3PbI_3$ NWs fabricated using Method-I and Method-II. (a) Dark-current of a polycrystalline NW device based on single NW. (b) Dark-current of a single-crystal NW device based on a single NW. (c) SEM image of a polycrystalline NW perpendicular to two Au electrodes. (d) SEM image of a single-crystal NW perpendicular to two Au electrodes. (e) Illustration of grain boundaries working as paths for iodine ion migration. (f) I and Pb distribution along the polycrystalline NW before and after poling different durations. (g) I and Pb ratios at anode and cathode after poling for 5 min. Each error bar was calculated from five independent polycrystalline NWs (average \pm std). (h) I and Pb ratio for 5 min. Each error bar was calculated for different durations. (i) I and Pb ratios at anode and cathode side after poling for 5 min. Each error bar was calculated from five independent polycrystal anode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode and cathode side after poling for 5 min. Each error bar was calculated from five independent strates at anode and cathode and cathode side after poling for 5 min.

Fig. 3e). The distribution of I : Pb along the GB-rich NW before and after electrical poling is depicted in Fig. 3f. After a moderate poling of 0.5 V μm^{-1} was applied for 3 min at 330 K, I : Pb dropped below 3:1 near the cathode but increases above 3:1 near the anode. This tendency becomes more obvious after 2 min of additional electrical poling and is mainly attributed to the I⁻ ionic migration and redistribution along the NW through the GBs. That is, I⁻ ionic migration in the presence of the electric field occurs not only at long GBs of large grains but also in the entire connected GB network of small grains. No obvious change of I: Pb was observed for the single-crystalline NW before and after poling under the same experimental conditions (Fig. 3h). This can be attributed to the lack of GBs to provide paths for ion migration in the high-quality single-crystalline CH₃NH₃PbI₃ NW. It was further confirmed by I and Pb ratios measured at the anode side and the cathode side of 5 polycrystalline NWs and 5 single-crystalline NWs after poling for 5 min (Fig. 3g and i). The quantitative analysis of ion migration was possibly limited by the relatively low accuracy of energydispersive X-ray spectroscopy (EDS). Still, the obtained data permits estimating the influence of GBs on ionic migration along a long NW. The result also shows that the singlecrystalline NWs made by Method-II are more suitable for photodetectors than GB-rich NWs prepared by Method-I.

We characterized the $CH_3NH_3PbI_3$ NWs produced by the micro-/nanofluidic fabrication platform for NW-integrated photodetectors. In Fig. S4,† the current-voltage (*I-V*) curves of

the as-prepared CH₃NH₃PbI₃ single-crystal NW arrays measured in the dark and under illumination are shown. We deposited two Au electrodes with a 100 µm spacing distance to connect the two ends of the NW arrays to complete a photodetector device integrated with a CH₃NH₃PbI₃ single-crystal NW array. After measuring dark currents, we repeated the same measurement for PCs by exposing the entire device to radiation at the wavelength of $\lambda = 532$ nm and the incident light intensity of 5 mW cm⁻². With the increase in the number of CH₃NH₃PbI₃ singlecrystal NWs from one to three, both the dark currents and PCs increase. The light absorption generates electron-hole pairs extracted by the source-to-drain electric field under illumination. This causes an increase in the material conductance by up to several dozen times.28 The increase from one to three CH₃NH₃PbI₃ single-crystal NWs in the photodetectors corresponds to an increase in photosensitivity from $\sim 1400\%$ to \sim 2900%. Increasing the number of NWs per device is easy with our fabrication method, so it is to increase photosensitivity. Photosensitivity can also be enhanced by decreasing the bridging-gap length of 100 µm, above which the recombination of electron-hole pairs is a factor. We investigated the darkcurrent hysteresis and ion migration along the long NW and used the bridging-gap length of 20 µm in the photodetectors we created from here.

In Fig. 4a, a schematic of a photodetector made with our method for highly sensitive, broadband photodetection is presented. The working mechanism of the photodetector is based



Fig. 4 Sensitive broadband photodetector integrated with ZnO and $CH_3NH_3PbI_3$ NWs in parallel. (a) Schematic of the photodetector work mechanism on two heterogeneous NWs. The ZnO NW detects UV light, $CH_3NH_3PbI_3$ visible light. (b) Schematic of the PDMS micro-/nanofluidic device with two separate nanochannels. On the right is the microscopic image of the micro-/nanofluidic device with two independent, parallel crack nanochannels. (c) SEM images of the parallel ZnO NW (left) and $CH_3NH_3PbI_3$ NW (right) on the same substrate and parallel to each other. (d) Dark and laser-illuminated *I*–*V* curves obtained from the ZnO NW under different illumination intensities of 360 nm light. (e) Temporal response with -1 V bias under 360 nm LED illumination with light intensities of 100 W cm⁻² and a modulation frequency of 0.2 Hz. (f) Dark and laser-illuminated *I*–*V* curves obtained from the CH₃NH₃PbI₃ NW under different illumination intensities of 532 nm and 360 nm light. (g) Temporal response with -4 V bias under 532 nm LED illumination with light intensities of 100 μ W cm⁻² and a modulation frequency of 250 Hz.

on two different NWs. The first ZnO NW detects UV light, and the second CH₃NH₃PbI₃ NW detects visible light, allowing the sensing of a wide range of wavelengths. We designed a microfluidic channel network and created crack nanochannels to grow two parallel and separate NWs at a designated location on the same substrate, as shown in Fig. 4b. This micro-/nanofluidic device is placed on a SiO₂/Si wafer, as shown in the microscopic image in Fig. 4b. A ZnO NW is first fabricated in the left crack nanochannel by a hydrothermal reaction²⁹ and then a CH₃-NH₃PbI₃ NW is prepared in the right crack nanochannel by using Method-II. The ZnO NW and CH₃NH₃PbI₃ NW are thereby successfully grown on the same substrate (Fig. 4c). The XRD pattern of the substrate displays the typical peaks of both ZnO and CH₃NH₃PbI₃ (see Fig. S5[†]). An enlarged SEM image and a cross-sectional SEM image of the ZnO NW can be seen in Fig. S6.† The ZnO NW consists of small particles. Au electrodes with bridging-gap lengths of 20 µm are deposited at the ends of the two NWs. As depicted in Fig. 4a, the electrode pairs are physically separated because ZnO and CH₃NH₃PbI₃ NW work under different bias voltages.

The single ZnO NW exhibited not only ~20 000% of the darkcurrent photosensitivity under UV light illumination at 1 mW cm⁻², as shown in Fig. 4d, but also a linear response from 20 μ W cm⁻² to 5 mW cm⁻² (Fig. S7a†). The ZnO NW response time, defined as the time necessary for the PC to rise from 10% to 90% of the maximum, and the recovery time, defined as the time to fall the same amount, are measured as ~0.6 s and ~0.9 s, respectively (Fig. 4e). These times indicate a higher photo response than observed in other photodetectors utilizing ZnO nanostructures.^{30,31} The high surface-area-to-volume ratio of ZnO NWs can contribute to O₂ re-adsorption³² as well as the generation of Schottky barriers³³ between metal electrodes.

Another key figure-of-merit for photodetectors is the responsivity (*R*), expressed as $R = J_{\rm ph}/L_{\rm light}$,³⁴ where $J_{\rm ph}$ is the PC density, and $L_{\rm light}$ is the incident light intensity. The PC density is given by $J_{\rm ph} = I_{\rm ph}/S$, where $I_{\rm ph}$ and *S* are the PC and effective device area, respectively. Accordingly, *R* for the single ZnO NW is estimated as 1727 A W⁻¹ under 360 nm light (1 mW cm⁻²) at the bias voltage of -1 V. The high PC ($I_{\rm ph} \approx 0.76 \ \mu$ A) and the small effective area ($S \approx 44 \ \mu$ m², around 20 μ m in length and 2.2 μ m in width) of the single NW device are intrinsic and unique advantages. The figure-of-merit is even higher than that of most ZnO-nanostructure-based photodetectors (see ESI Table S1†).^{30,31,35,36}

In Fig. 4f, the photosensitivity of the CH₃NH₃PbI₃ NW is shown that reaches \sim 6300% under 532 nm laser illumination at 1 mW cm⁻² compared to the dark-current. Similarly, the CH₃- NH_3PbI_3 single NW responds linearly from 50 μ W cm⁻² to 50 mW cm $^{-2}$ (Fig. S7b[†]). The response and recovery times of the single CH₃NH₃PbI₃ NW are ~0.28 ms and 0.92 ms, respectively (Fig. 4g), comparable to other photodetectors utilizing perovskite NWs (Table S1[†]).^{5,6,28} The *R* value for a single CH₃NH₃PbI₃ NW is estimated at 275 A W⁻¹ ($I_{\rm ph} \approx 0.11 \,\mu\text{A}$ and $S \approx 40 \,\mu\text{m}^2$, 20 μ m in length and 2 μ m in width) under 532 nm light (1 mW cm^{-2}) at the bias voltage of -1 V, which is also impressively high.^{5,6,37} The *R* value of a single CH₃NH₃PbI₃ NW is about 85 A W^{-1} ($I_{ph} \approx 0.034 \,\mu\text{A}$ and $S \approx 40 \,\mu\text{m}^2$, 20 μm in length and 2 μm in width) under 360 nm light (1 mW cm $^{-2}$) at the bias voltage of -1 V. Notably, the photosensitivity of ZnO NWs is greater than that of CH₃NH₃PbI₃ NWs by a factor of 20 under the same UV

illumination conditions, thereby overpowering the photosensitivity of CH₃NH₃PbI₃ NWs and clearly demonstrating the synergistic advantages of combining two different semiconducting NWs in an array.

We reported in our previous work¹⁹ that crack photolithography can be used to manipulate the number, dimension, orientation, and alignment of nanoscale cracks on an SU-8 master mold. Such cracks can be recreated as a PDMS device via poly (urethane acrylate) (PUA) replication (i.e., as the second mold for PDMS). Our method easily controls important factors using standard photolithography, without nanofabrication tools and equipment. Such controllability may not be achievable in conventional nanofabrication. The limitation of this method is that only one kind of nanowire can be synthesized in turn. But our device overcomes the problem of mutual contamination in the synthesis of different NWs. Our micro-/ nanofluidic fabrication method may allow a batch process for other NWs made of other materials as it can be applied to any solution processing (e.g., the syntheses for ZnS,³⁸ TiO₂,³⁹ SnO₂ (ref. 40) or metals such as Au,⁴¹ Ag,⁴² and Cu⁴³). Multi-functional detectors combining differently functionalized NW sensors such as gas- and light-sensitive NWs were made.9 We, therefore, have demonstrated that our method facilitates the synthesis, alignment, and operation of NWs on various substrates in a simple, low-cost, labor-reduced, and repeatable manner.

Conclusions

We demonstrated a new NW fabrication method that produces NWs with different materials or NWs made of the same materials but exhibiting different morphologies using the same micro-/ nanofluidic platform by solution process. We fabricated two types of CH₃NH₃PbI₃ NWs with varying crystalline properties by GB richness and single crystallinity under varied processing conditions. The CH₃NH₃PbI₃ NWs find use in photodetectors and allow an ideal material research platform to study ion migration. The proposed fabrication method enables the production of different NWs, such as ZnO NWs and CH₃NH₃PbI₃ NWs, on the same substrate. Both solid and flexible substrate materials are permitted by using a PDMS micro-/nanofluidic device for solution loading. We demonstrated a wide-range photodetector with relatively high sensitivity to both UV light with the ZnO NW and visible light with the CH₃NH₃PbI₃ NW made with a conventional nanofabrication process. We demonstrated only some of the capabilities of our proposed method in developing the photodetector nanosystem but thoroughly anticipate the wise spread use of our method for low-cost, large-scale, highly efficient, and flexible optoelectronic applications. Our proposed fabrication method, which allows for the control of the number, dimension, orientation, alignment, position, and material of NWs, will facilitate a variety of NW applications in the near future.

Methods

Reagents and materials

A Sylgard 184 silicone elastomer kit (Dow Corning, Midland, MI, U.S.A.) was used for PDMS-based soft lithography. A

polyurethane acrylate (PUA) solution, MINS-311RM, and a polyethylene terephthalate (PET) film (Minuta Tech., Osan, Korea) were used for replication of the SU-8 (SU-8 2010, MicroChem, Westborough, MA, U.S.A.) master mold. Lead(π) iodide (PbI₂), methylammonium iodide (CH₃NH₂·HI), hexamethylenetetramine (HMTA), zinc nitrate hexahydrate (Zn(NO₃)₂·6H₂O), isopropanol (IPA), and *N*,*N*-dimethylformamide (DMF) were purchased from Sigma Aldrich (St. Louis, MO, U.S.A.). CH₃NH₃PbI₃ powder (CAS: 69507-98-8) was purchased from Xi'an polymer Light Technology Corp., Xi'an, Shaanxi, China. All reagents were of analytical grade and used without further purification.

Fabrication of NWs using a micro-/nanofluidic platform with crack nanochannels

Two-step and open-boundary vapor deposition method (Method-I): 2 μ L PbI₂ DMF solution with the concentration of 0.25 M was dripped into the microchannels. After 30 min, the solution fully had filled the microchannels and the crack nanochannels by capillarity. The micro-/nanofluidic device was then heated to 50 °C and held for 30 min to evaporate the DMF. The micro-/nanofluidic device was further heated to 100 °C and held for another 10 min to altogether remove any DMF residues. The PDMS device was peeled off and removed from the SiO₂/Si substrate, and then the PbI₂ NWs samples were placed into a container. The container subsequently was moved into a furnace with CH₃NH₃I powder surrounding the substrate. Eventually, the PbI₂ NWs on the substrate were converted to CH₃NH₃PbI₃ NWs at 140 °C for 2 h.

One-step and closed-boundary crystallization method (Method-II)

 $2 \ \mu L$ as-prepared CH₃NH₃PbI₃ solution (0.5 M) was injected into the inlets of one side of two parallel PDMS microchannels. Then, the micro-/nanofluidic device was relocated into a glass container (Lock & Lock, USA) with a continuous flow of nitrogen gas as a protective gas. The glass container was placed on a hot plate and kept at 75 °C for 2 h. Lastly, the NW samples were additionally annealed at 100 °C for 1 h.

ZnO NW fabrication method

 $Zn(NO_3)_2$ and hexamethylenetetramine (HMTA) solutions at the concentration of 0.0025 mol L⁻¹, respectively, were mixed to a molar ratio of 1:1. The mixture was injected into the two microchannels sandwiching one crack nanochannel. The micro-/ nanofluidic device was then put on a hot plate at 90 °C and annealed for 2 h. The reaction was performed in a container in high humidity to limit the evaporation of the solutions through the PDMS device. During the entire process, the inlets and outlets of the microchannels were protected by tape to block any contamination.

Characterization and measurements of photo response of NWs

The growth process was recorded at a rate of 1 frame per 6 seconds from the beginning of injection of $CH_3NH_3PbI_3$

solution to monitor the entire growth of CH₃NH₃PbI₃ NW at 25 °C. The structural properties of our fabricated NWs were evaluated with AFM, DI-3100 (Veeco, New York, NY, U.S.A.), high resolution transmission electron microscopy (HR-TEM, JEM2100F, JEOL, Tokyo, Japan), X-ray diffraction (XRD, D8 Advance, Bruker AXS, Billerica, MA, U.S.A.), and field effect SEM (FE-SEM, S-4800, Hitachi, Tokyo, Japan), and EDS, (EMAX, Mahwah, NJ, U.S.A.). Electrical measurements of the photodetector nanosystems were conducted with a semiconductor characterization system (Model 4200, Keithley, Cleveland, OH, U.S.A.). Monochromatic light in the UV-vis range was obtained by filtering the white light from a high-power xenon lamp with optical filters, and a light intensitometer was used to determine the light intensity. A home-built measurement system that combined a laser diode (360 nm, 532 nm) and a pulse generator was used to quantify the photo response time of the photodetector nanosystems under differently pulsed light illumination conditions.

Conflicts of interest

There are no conflicts to declare.

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