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# First-principles study of two dimensional C<sub>3</sub>N and its derivatives†

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Here we have performed a comprehensive first-principles study for electronic and mechanical properties of newly synthesized C<sub>3</sub>N and its derivatives. The C<sub>3</sub>N monolayer is evaluated to be an indirect semiconductor with a HSE06 level bandgap of 1.09 eV, which can be effectively tuned by the number of layers, stacking order and B-doping concentration. With strong polar covalent bonds, C<sub>3</sub>N is predicted to be a superior stiff material with high in-plane Young's modulus (1090.0 GPa) and thermal dynamic stability (up to 2000 K). Remarkably, the C<sub>3</sub>N monolayer possesses a fascinating bending Poisson's effect, namely, bending induced lateral contraction, which is rare in other 2D materials. What's more, C<sub>3</sub>N nanosheets can be rolled into nanotubes with a tunable bandgap corresponding to the radius of curvature. Due to high stability, suitable band gap and superior mechanical strength, two dimensional C<sub>3</sub>N will be an ideal candidate in high-strength nano-electronic device applications.

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## 1 Introduction

Since the first two-dimensional (2D) material, namely graphene, was successfully fabricated by Novoselov and Geim,<sup>1</sup> 2D materials research has rapidly risen to be one of the hot spots of condensed matter physics due to their fascinating electronic, mechanical, optical or thermal properties.<sup>2–4</sup> For instance, many fantastic phenomena are discovered in graphene,<sup>5</sup> due to the presence of a Dirac-type band dispersion. However, the gapless feature, at the same time, also limits its application in electronic and optoelectronic devices. Subsequently, few layer black phosphorus<sup>6–8</sup> was successfully synthesized and it is predicted that it not only has a direct band gap of about 2.0 eV, but also has a high carrier mobility. Unfortunately, black phosphorene degrades readily when it exposes to the moist air.<sup>9</sup> At the same time, due to the low Young's modulus, the black phosphorene is too flexible to be applied under high mechanical environment.<sup>10</sup> Therefore, exploring new 2D semiconducting materials with high stability, strong mechanical strength and suitable band gaps are still a long-term target.

Recent years, a hole-free 2D crystal consisted of carbon and nitrogen atoms, named C<sub>3</sub>N, has been fabricated by polymerization of 2,3-diaminophenazine.<sup>11</sup> C<sub>3</sub>N monolayer possesses an indirect band gap of 0.39 eV (PBE level) that can be tuned to cover the entire visible range by fabrication of quantum dots with different diameters. The back-gated field-effect transistors made of monolayer C<sub>3</sub>N display a high on/off ratio of  $5.5 \times 10^5$ .

More importantly, an experimental research designs an artificial synapse with tunable synaptic behavior based on solution-processed 2D C<sub>3</sub>N/polyvinylpyrrolidone (PVPy), which can mimic the synapse cleft based on proton conducting mechanism and may find further applications in artificial intelligence.<sup>12</sup> Therefore, a further examining of the mechanical and electronic structure properties will be helpful to further exploit other applications of this new type of 2D material.

In this work, we firstly present the basic methodology that we used to in this simulation. Then detailed results of geometry structure, and electronic properties of layered C<sub>3</sub>N are discussed, including the mechanical properties of two-dimensional C<sub>3</sub>N and C<sub>3</sub>N-based nanotubes. Finally, we give conclusions and outlooks.

## 2 Computational methods

In this work, all the first-principles calculations are performed based on the Kohn–Sham density functional theory<sup>13</sup> (KS-DFT) as implemented in the Vienna *Ab initio* Simulation Package<sup>14</sup> (VASP). The generalized gradient approximation within the Perdew–Burke–Ernzerhof<sup>15</sup> (PBE) functional form is used for the exchange–correlation energy. The plane wave basis sets with kinetic energy cutoff of 500 eV are used to expand the valence electron wave functions. For all structural relaxations, the convergence criterion for the energy in electronic SCF iterations and the Hellmann–Feynman force in ionic step iterations are set to be  $1.0 \times 10^{-6}$  eV and  $5.0 \times 10^{-3}$  eV Å<sup>-1</sup>, respectively. In order to reduce the interaction between neighboring layers, a large vacuum space of at least 15 Å is introduced along the z-axis. The Brillouin zone is represented by Monkhorst–Pack<sup>16</sup> special *k*-point mesh of  $12 \times 12 \times 1$  for geometry optimizations,

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while a larger grid ( $16 \times 16 \times 1$ ) is used for SCF computations. Besides, HSE06 hybrid functional is used to obtain an accurate band gap.<sup>17</sup> van der Waals (vdW) correction proposed by Grimme (DFT-D2) is used due to its good description of long-range vdW interactions for multi-layered 2D materials.<sup>18–22</sup> As a comparison, the Becke88 van der Waals<sup>23</sup> (optB88-vdW) functional is also used for multi-layered structure. *Ab initio* molecular dynamics (AIMD) simulations with NVT ensemble are performed to assess the thermal stability of  $C_3N$  monolayer.

### 3 Results and discussion

#### 3.1 Geometry structure and electronic properties of layered $C_3N$

The optimized structure of  $C_3N$  monolayer is shown in Fig. 1(a). The structure possesses  $P6/mmm$  symmetry (space group ID 191) with hexagonal lattice. The optimized lattice constants are  $a = b = 4.862 \text{ \AA}$ . The top view shows that the new phase is similar to graphene sheet, however, the six membered rings are composed by either C and N atoms or all C atoms. Unlike blue phosphorene,<sup>24</sup> silicene<sup>25</sup> and some other puckered materials,<sup>26,27</sup> the  $C_3N$  has a flat structure from the side view. In this structure, all C and N atoms are  $sp^2$  hybridized forming conjugated  $\pi$  bond. The C–C ( $1.40 \text{ \AA}$ ) and C–N ( $1.40 \text{ \AA}$ ) bond lengths show pronounced characters of single bonds. The unit cell of  $C_3N$  contains 8 atoms as denoted by red box in Fig. 1(a) in which the C to N ratio is 3 : 1.

Before studying the electronic and mechanical properties of  $C_3N$ , we firstly determine whether it is stable or not. To confirm the dynamical stability, the phonon dispersions of  $C_3N$  are calculated by using the finite displacement method as implemented in PHONOPY.<sup>28</sup> The calculated phonon dispersion curve in Fig. 1(b) shows no imaginary modes in the entire Brillouin zone, which confirms that  $C_3N$  is dynamically stable. Moreover, we also carried out *ab initio* molecular dynamics (AIMD) simulation with  $4 \times 4 \times 1$  supercell to judge the thermodynamic stability of  $C_3N$  monolayer. After heating at high temperature (1000 K) for 5 ps with a time step of 1 fs, no structure reconstruction is found. Furthermore study indicates that  $C_3N$  can also withstand temperature under 2000 K. Remarkably, the melting point is getting close to 3000 K, suggesting that  $C_3N$  monolayer has superior thermal stability (see in ESI Fig. S1†).

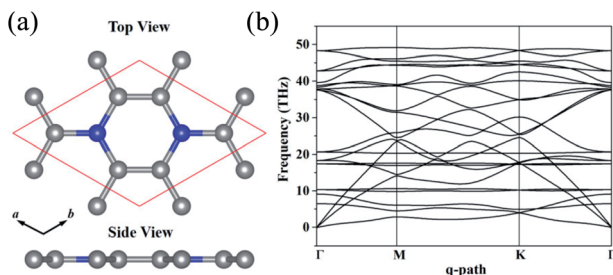


Fig. 1 (a) Top and side view of monolayer  $C_3N$  in the unit cell. (b) Phonon band structure  $C_3N$ .

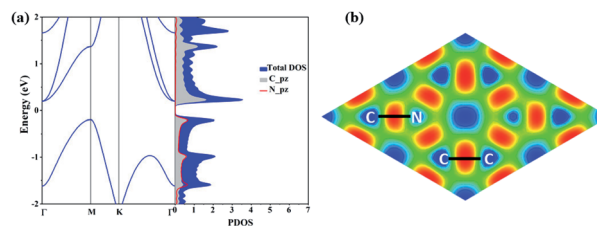


Fig. 2 (a) Electronic band structure and projected density of state (PDOS).  $\Gamma$  (0.0, 0.0, 0.0), M ( $1/2, 0.0, 0.0$ ) and K ( $-1/3, 2/3, 0.0$ ) refer to special points in the first Brillouin zone. (b) The electron localisation function (ELF) of  $C_3N$ .

The electronic structure properties of  $C_3N$  are then investigated. The electronic band structure as well as density of states (DOS) of  $C_3N$  monolayer are calculated under the PBE level of theory. As shown in Fig. 2(a), unlike gapless graphene,<sup>29</sup>  $C_3N$  is predicted to be indirect band gap semiconductor with a gap of 0.39 eV, as the valence band maximum (VBM) and the conduction band minimum (CBM) are located at the M point and the  $\Gamma$  point in the Brillouin zone, respectively. From the projected band structure and density of states (DOS), one can see that the VBM is a hybrid state of C-pz and N-pz, however, the CBM is mainly contributed by pz orbital of C atoms, which is consistent with their electronic charge density (see in ESI Fig. S2†). The calculated bandgap value is consistent with

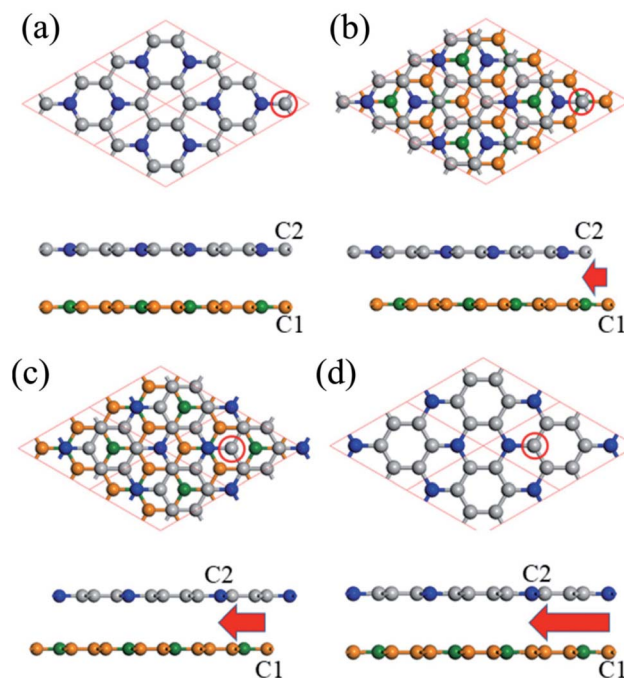


Fig. 3 Four stacking structures of bilayer  $C_3N$ , (a)–(d) top view and side view of AA-, AB-, AC- and AD-stacking, respectively. A  $2 \times 2$  supercell is adopted for the top view. The red arrow indicates the relative shift of C1 and C2 atom in the top layer is labelled by red circle. The C atoms in different layers are denoted by gray and yellow spheres. And the N atoms in different layers are denoted by blue and green spheres.



Table 1 The relative energy, layer distance and bandgap of bilayer C<sub>3</sub>N under different functional

Stacking order	Relative energy eV per unit cell		Layer distance Å		Bandgap eV		
	DFT-D2	optB88	DFT-D2	optB88	DFT-D2	optB88	HSE06
AA	0.093	0.086	3.40	3.41	0.00	0.00	0.27
AB	0.003	0.007	3.22	3.25	0.22	0.23	0.83
AC	0.017	0.017	3.20	3.20	0.01	0.08	0.62
AD	0.000	0.000	3.23	3.21	0.00	0.00	0.48

previous work.<sup>11</sup> It is well-known that DFT within PBE level of theory underestimates the bandgap of semiconductors.<sup>30</sup> Therefore, a more accurate hybrid functional HSE06 (ref. 17) is employed to correct the bandgap. We verify that the band dispersion profiles remain the same (see in ESI Fig. S2†), but the band gap value increases to 1.09 eV. By calculating the electron localization function (ELF),<sup>31,32</sup> the C–C and C–N covalent bonds in C<sub>3</sub>N can be identified (Fig. 2(b)).

In addition, according to our Bader's charge analysis,<sup>33</sup> each C atom transfers about 1.3e to N atoms, which implies a significant polarizability of C–N covalent bonds. Compared with previously proposed N chains encapsulated in carbon nanotubes<sup>34</sup> (0.4e) and the penta-CN<sub>2</sub> (ref. 35) (1.2e), the nitrogen atoms in C<sub>3</sub>N receive much more charge, indicating a much stronger interaction between C and N atoms, which to some extent indicates that C<sub>3</sub>N will hold high Young's modulus.

We also calculate the multi-layered C<sub>3</sub>N nanosheet. As for the geometric structure of bilayer C<sub>3</sub>N, we here consider four kinds of high symmetry stacking order, namely AA-, AB-, AC- and AD-stacking. As shown in Fig. 3a, in AA-stacking, the top layer is directly stacked on the bottom layer and they are matched perfectly in *xy*-plane. The AB-, AC and AD-stacking can be viewed as shifting the top layer of atoms along the vector *a*-*b* with a displacement as displayed in Fig. 3. As a comparison, the relative energy, layer distance and bandgap are listed in Table 1. We firstly optimize the structure by including DFT-D2 correction due to its good description of interlayer interaction. The calculated layer distance is in the range of 3.20–3.40 Å, which is close to the value of its parent material graphene (3.35 Å)<sup>36</sup> and analogue C<sub>2</sub>N-h2D (3.18 Å).<sup>37</sup> By analyzing the relative energy, we may find that AD-stacking is the most favorable configuration for bilayer C<sub>3</sub>N, being 0.093, 0.03 and 0.017 eV per unit cell lower than that of AA-, AB and AC-stacking, respectively.

In addition, the calculated relative energy and layer distance with optB88 vdW functional are consistent with the DFT-D2 ones, this further demonstrates the validity of our calculation. Due to the small energy difference, these configurations are expected to transform to each other under appropriate condition. Therefore, it is desired to investigate electronic structure of different stacking order. As listed in Table 1, the most stable configuration AD-stacking and metastable AA-stacking, the AB- and AC-stacking are predicted to be semiconductors with a HSE06 level bandgap of 0.48, 0.27, 0.83 and 0.62 eV, respectively. For trilayer C<sub>3</sub>N, we consider four kinds of high symmetry

stacking structures based on the three stacking orders of bilayer C<sub>3</sub>N: ADA-stacking, ADB-stacking, ADC-stacking, and ADD-stacking. The results show that ADA-stacking is the most stacking order. On the whole, the electronic structure of multi-layered C<sub>3</sub>N is closely related to the number of layers and stacking order. The bandgap can be effectively tuned by the number of layers and stacking order, which supplies an optional way to tune electronic structure of C<sub>3</sub>N.

Doping has been widely used to tune the electronic structures of 2D materials.<sup>38,39</sup> In this work, we systematically study the electronic structures of bilayer C<sub>3</sub>N by B doping. Here we only consider the most stable configuration (AD-stacking) for B doping. A 2 × 2 × 1 supercell of bilayer C<sub>3</sub>N is adopted to analyze the B doping effects. There are eight N atoms in each layer of C<sub>3</sub>N. A hexagonal pattern of N atoms are substituted by the B dopant atoms, including two series of configurations. (i) Only one layer of bilayer C<sub>3</sub>N is doped with 1, 2, 3 and 4 B-atoms, corresponding to dopant concentrations of 6.25%, 12.5%, 18.75% and 25%, respectively (Fig. 4). (ii) Both layer of C<sub>3</sub>N is doped with 1, 2, 3 and 4 B-atoms, corresponding to dopant concentrations of 12.5%, 25%, 37.5% and 50%, respectively. Based on the PBE-level simulation with DFT-D2 correction, the band structures of bilayer B-doped C<sub>3</sub>N are obtained in Fig. S3.† For case (i), B-doped bilayer C<sub>3</sub>N with concentration of 12.5% is metal. The B-doped bilayer C<sub>3</sub>N with concentration of 6.25% is an indirect semiconductor with a bandgap of 0.023 eV, while 18.75% and 25% are direct semiconductors with a bandgap of 0.15 eV and 0.24 eV, respectively. For case (ii), B-doped bilayer C<sub>3</sub>N with concentration of 12.5% is metal. The B-doped bilayer C<sub>3</sub>N with concentrations of 25%, 37.5% and 50% are indirect semiconductors with a bandgap of 0.21 eV, 0.48 eV and 1.30 eV,

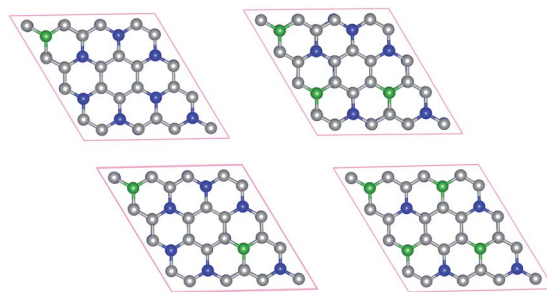


Fig. 4 Structures of B doped bilayer C<sub>3</sub>N with 1, 2, 3 and 4 B-atoms for (i) case. The C atoms, N atoms and B atoms are denoted by gray, blue and green spheres.



respectively. In all, the bandgap of AD-stacking  $C_3N$  can be tuned by B doping. Particularly, in case (ii), the bandgap of AD-stacking  $C_3N$  increases as the doping rate increases, which makes  $C_3N$  suitable for designing modern electronic devices.

### 3.2 Mechanical properties and $C_3N$ nanotubes

By using the finite distortion method,<sup>40</sup> the linear elastic constants of  $C_3N$  monolayer are obtained (see in ESI Table S1†). Due to the symmetry of geometric structure, there are three independent elastic constants for 2D hexagonal crystal, namely,  $C_{11}$  ( $C_{22} = C_{11}$ ),  $C_{12}$  and  $C_{66}$ . The elastic constants of  $C_3N$  satisfy the requirement of the Born criteria,<sup>41</sup> namely  $C_{11} > 0$ ,  $C_{66} > 0$  and  $C_{11} - C_{12} > 0$ , which further demonstrates the mechanical stability of  $C_3N$ . To have a deep knowledge of mechanical properties of  $C_3N$ , we plot the orientation dependent Young's modulus and the Poisson's ratio according to eqn (S1) (see in ESI).† As comparison, the Young's modulus and the Poisson's ratio of some representative 2D materials, such as graphene<sup>42,43</sup> and penta-CN<sub>2</sub>,<sup>35</sup> are also plotted Fig. 5(a) and (b). Unlike penta-CN<sub>2</sub>, the Young's modulus and the Poisson's ratio of  $C_3N$  is isotropic due to its high symmetry. More important than the isotropic mechanical properties, the Young's modulus of  $C_3N$  (1090.0 GPa) is comparable or a little higher than that of graphene (1057.7 GPa) and much higher than that of penta-CN<sub>2</sub> (794.7 GPa), borophene (646.6 GPa)<sup>44</sup> and black phosphorene (166.0 GPa),<sup>10</sup> which suggests that 2D  $C_3N$  is as stiff as graphene. To explore the ideally tensile strength (the highest achievable stress of a defect-free crystal at 0 K) and critical strain (the strain at which ideal strength reaches)<sup>45</sup> of  $C_3N$ , an in-plane biaxial tensile stress is applied. The stress-strain relationship for monolayer  $C_3N$  is presented in Fig. S4 (see in ESI),† where the tensile strain ranges from 0 to 20%. The ideal strength under the critical strain (14%) is 207.1 GPa (equivalent value 66.5 N m<sup>-1</sup>), which is apparently larger than that of black phosphorene,<sup>10</sup>  $\delta$ -phosphorene,<sup>46</sup> MoS<sub>2</sub> (ref. 47) and borophene.<sup>48</sup> From the high Young's modulus and large ideal strength, we may conclude that  $C_3N$  monolayer is a stiff material.

To simulate the out-of-plane bending deformation of  $C_3N$  monolayer, the  $C_3N$  nanosheet is rolled into the corresponding nanotube, which has been widely studied in other 2D materials, such as graphene,<sup>49,50</sup> boron nitride,<sup>51</sup> blue phosphorene<sup>52</sup> and

black phosphorene.<sup>53</sup> Here, monolayer  $C_3N$  is directly rolled up to  $C_3N$ -nanotube. Specifically, two types of nanotube is considered in this work, namely, zigzag ( $n, 0$ ) for  $n = 3, 4, \dots, 7$  and armchair ( $n, n$ ) for  $n = 2, 3, \dots, 6$  (see in ESI Fig. S5†).

According to the Poisson's ratio formulates, lateral strains in a material can be caused by a uniaxial stress in the perpendicular direction, but no net lateral strain should be induced in a thin homogeneous elastic plate subjected to a pure bending load. Here, we find that significant exotic lateral strains can be induced while bending  $C_3N$  sheet to corresponding nanotube (see Fig. 6(a)). Taking zigzag nanotube as an example, the lattice constant  $r_0$  of all ( $n, 0$ ) tube should be equal to the width of  $C_3N$  sheet without bending load  $r_d$ . However, as listed in Table S2,† all of lattice constants  $r_0$  are smaller than  $r_d = 8.422 \text{ \AA}$ , indicating that  $C_3N$  monolayer has an interesting bending Poisson effect.<sup>54</sup> To have deep knowledge of this behavior, the bending Poisson ratio  $\nu$ , defined as the ratio of lateral strain to the curvature of  $C_3N$ -nanotube ( $\nu = \varepsilon_L/k$ ) is evaluated, where the lateral strain introduced by the bending load can be written as  $\varepsilon_L = (r_d - r_0)/r_0$  and corresponding lateral stress is defined as  $\sigma_L = Y\varepsilon_L$ .  $Y$  is the in-plane Young's modulus of  $C_3N$  monolayer. According to the  $\varepsilon_L$ - $k$ ,  $\sigma_L$ - $k$  and  $\nu$ - $k$  relations illustrated in Fig. 6(b)–(d), some typical features of the bending Poisson's effect in  $C_3N$  monolayer can be summarized: (i) the bending-induced lateral strain is orientation-dependent. For instance, when the radius of curvature down to 2.6  $\text{\AA}$ , the lateral strain along the armchair direction is  $-0.45\%$ , while the zigzag direction is up to  $-1.45\%$ , which should be easy to be detected by experiment. In addition, the zigzag nanotubes with small radius of curvature have a more obvious bending Poisson's effect than the armchair ones, however, with the increase of

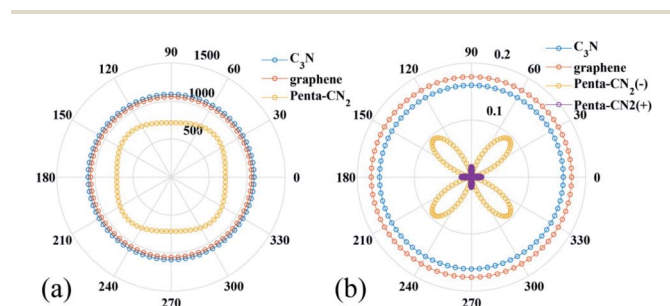


Fig. 5 (a) Orientation dependent (a) Young's modulus (GPa) and (b) Poisson's ratio of  $C_3N$ , penta-CN<sub>2</sub> and graphene. Penta-CN<sub>2</sub>(-) and penta-CN<sub>2</sub>(+) indicate negative and positive Poisson's ratio of penta-CN<sub>2</sub>, respectively.

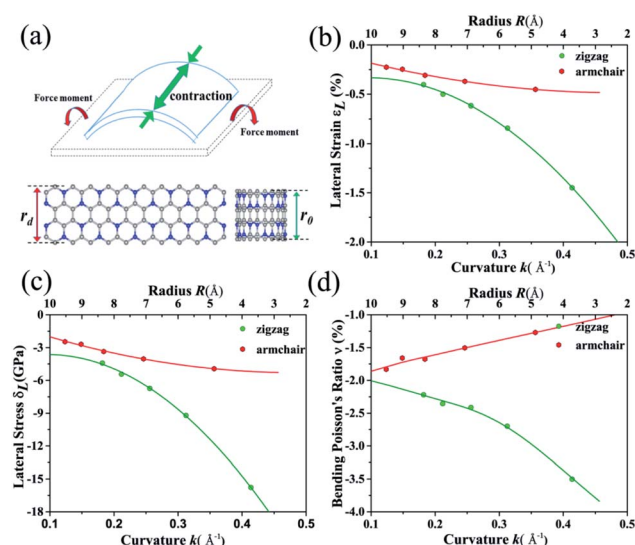


Fig. 6 (a) Schematic plot of homogeneous elastic plate under a pure bending loading. The plates with and without mechanical load are depicted by the blue and black dashed line.  $C_3N$  nanosheet and corresponding nanotube are also included. (b), (c) and (d) Bending induced lateral strain, equivalent lateral stress and bending Poisson's ratio as a function of curvature for bending along the zigzag (green line) and armchair (red line) directions.



radius of curvature, the lateral strain of both armchair and zigzag nanotubes reduces to the 0. (ii) The variation trend of lateral stress under different radius of curvature is similar to that of lateral strain. Compared with in-plane strain–stress curve of  $C_3N$  monolayer, the lateral stress is much smaller than in-plane strain-induced equivalent stress. (iii) The bending Poisson ratio is a function of curvature. As for armchair nanotube, the  $\nu$ – $k$  relation is approximately linear, while zigzag nanotube has a nonlinear relation. To explain bending Poisson's effect, a local structure of  $C_3N$  nanotube is shown in Fig. S6 (see in ESI†). It can be seen that the perfect  $C_3N$  sheet has a flat structure, where all carbon and nitrogen atoms hold  $sp^2$  hybridization. However, when out-of-plane bending load is employed to the  $C_3N$  monolayer, the  $sp^2$  hybridization states are destroyed. To accommodate the bending load, the N atoms will be prone to form  $sp^3$  hybridization as it is in  $NH_3$  molecule. Finally, the C atoms linked the same N atom will get close to each other and lead to an axial contraction.

Finally, as a byproduct of bending effect investigation, the basic electronic and mechanical properties of  $C_3N$  nanotubes are also investigated (see in ESI Table S2†). For both armchair and zigzag  $C_3N$  nanotube, as expected, the strain energy decreases while  $d$  increases. The Young's modulus of all the nanotubes we selected are approximately equal to 1000.0 GPa except the (3, 0)-nanotube, which are consistent with recent work.<sup>55,56</sup> It is worth noting that the Young's modulus of tubes is also comparable or a little higher than that of graphene. A qualitative calculation shows that all of them are semiconductors with their bandgaps related with radius of curvature  $d$ . However, the zigzag nanotubes give rise to a much smaller band gap compared to that of the armchair nanotubes with nearly the same  $d$ . In detail, the band gap of (5,0)-nanotube is 0.97 eV, while that of (3, 3)-nanotube is up to 1.43 eV, which is even higher than that of flat  $C_3N$  monolayer, indicating that bending load can be a tool to tune the band gap of  $C_3N$ . In addition, we calculate the optical absorption for  $C_3N$  nanotubes as displayed in Fig. S7.†  $C_3N$  nanotubes show absorption in the visible-light and UV-light range. What's more, we can modulate the energy range of the absorption peak with different tubes. The absorption peak of the armchair tube always has a small blue shift relative to the zigzag tube. Considering the excellent properties of optical absorption for  $C_3N$  nanotube, it is suitable for optoelectronic device.

## 4 Conclusions

In summary, we conduct a first-principles simulation to investigate the thermal and dynamic stability, electronic and mechanical properties of two dimensional  $C_3N$  and its derivatives.  $C_3N$  monolayer is predicted to be a semiconductor with suitable bandgap, ultra-high mechanical strength and thermal stability, whose electronic properties can be tuned by layer number and stacking order. As for mechanical aspect,  $C_3N$  possesses a fascinating bending Poisson's effect, resulting from the change of hybridization state of N atoms in local  $C_3N$  skeleton. Besides, most of the corresponding nanotubes also present high Young's modulus and semiconducting properties,

which may extend the application of  $C_3N$  materials. Considering these high stability, superior mechanical strength and semiconducting properties of  $C_3N$  and related derivative, two dimensional  $C_3N$  is expected to have promising potentials to compete not only against graphene but also against other 2D materials for various applications, particularly in nanotransistors, fabrication of polymer nanocomposites with superior mechanical response. We hope our research can stimulate more experiments work on this subject.

## Conflicts of interest

There are no conflicts of interest to declare.

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