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Advanced development of metal oxide nanomaterials for H₂ gas sensing applications

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Hydrogen (H₂) has been considered as one of the cleanest renewable energy sources. However, it is still challenging to use H₂ due to its hazardous flammable and explosive properties under mild conditions in the event of leakage, and the difficulty to detect or sense it through human sensory organs because of its colorless and odorless nature. Traditional detection methods are usually complicated and the testing instruments are expensive. Thus, it is of significant importance to develop sensors for H₂ detection with facile operation conditions, low costs, and excellent performance (*i.e.*, sensitivity, selectivity, and stability). To overcome the problems and for practically detecting H₂ gas, metal oxide (MOx) nanomaterials have become more crucial in such a gas sensor because of the simple preparation method, high surface area, high sensitivity, and low costs. This review will focus on the recent state-of-the-art advances in resistive H₂ gas sensors based on MOx nanomaterials, starting from a brief introduction of resistive gas sensors. The following sections will focus on the synthesis of different structures and types of such MOx nanomaterials, including mono/binary/ternary/ternary or more complicated MOx nanomaterials. Meanwhile, we highlight some regulation methods such as surface or inner decoration by noble or non-noble metals to improve the performance as well as summarize and compare different structures (core-shell and heterojunction), and mechanisms in H₂ sensing. Finally, the opportunities and challenges of MOx-based H₂ gas sensors are proposed in detail.

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1. Introduction

Energy plays a crucial role in promoting the development of human society. With the severity of environmental pollution and the increasing consumption of non-renewable resources, it is urgent to find clean energies as an alternative to fossil fuels. Among them, hydrogen is one of the most ideal choices.^{1,2} However, H₂ has a low explosion limit (~4%) and a wide explosion range (4–75%) in air.³ In addition, it may lead to hypoxic asphyxia with the accumulation of H₂ in air.⁴ In order to detect H₂ leakage during storage and transportation in time and to avoid large-scale social hazards, various H₂ detection devices have been developed.^{5–9} Of achievements so far, resistive gas sensors are most favored by researchers.^{10–12}

Semiconductor metal oxide (MOx) nanomaterials have extraordinary physical-chemical properties in optical, electric, and magnetic performances,^{13–15} which have been widely used in catalysis,^{16–22} energy storage,²³ biosensors,²⁴ and gas sensors.^{25–29} There are a large number of active sites on the surface of MOx, which is beneficial for the adsorption of gases and the occurrence of chemical reactions.^{30–32} In particular, they are the key to the

construction of gas-sensing platforms. Resistive H₂ gas sensors based on MOx nanomaterials have been widely studied.^{33–38} A series of MOx nanomaterials with different morphologies and structures have been prepared by a variety of methods for the rapid and efficient detection of H₂.^{39–43} With the development of nanotechnology, the response is much improved.⁴⁴ However, the low selectivity, poor stability, and weak durability of H₂ gas sensors still impede their practical applications.^{45–51}

Although several reviews related to H₂ gas sensors have been published, if one is focused on chromic H₂ gas sensors,⁵² others are focused on a single or a class of MOx nanomaterials.^{53,54} For example, Luo *et al.*⁵³ and Mirzaei *et al.*⁵⁴ discussed the applications of noble-doped MOx nanomaterials in H₂ gas sensing. Li *et al.*³³ gave a comprehensive summary of resistive H₂ gas sensors based on TiO₂ and a review by Ren *et al.*³⁵ covered ZnO resistive H₂ sensors. However, there is a lack of reviews that sum up MOx-based H₂ gas sensors from synthesis to application comprehensively.

In this review, we summarize the recent advances in the preparation strategies of MOx and the state-of-the-art applications of these nanomaterials in H₂ gas sensors comprehensively. First of all, we give a brief introduction about the related information of MOx nanomaterials gas sensors, including structure, parameters, and mechanism of H₂ gas sensors. Then,

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Table 1 Metal oxide nanomaterials structure for H₂ gas sensors

Category		Materials
Mono-semiconductor metal oxide nanomaterial	N type P type	In ₂ O ₃ , ¹⁵ WO ₃ , ⁴⁰ MoO ₃ , ⁵⁵ SnO ₂ , ⁵⁶ TiO ₂ , ⁵⁷ ZnO ⁵⁸
Metal@MOx nanocomposites	Noble metal decorating	CuO, ⁵⁹ NiO, ⁶⁰ TeO ₂ ⁶¹ Pt-In ₂ O ₃ , ⁶² Pt-Nb ₂ O ₅ , ⁶³ Pd-ZnO, ⁴⁵ Pt-NiO, ⁴⁷ Ag-ZnO, ⁶⁴ Au-In ₂ O ₃ , ⁶⁵ Au-ZnO, ⁶⁶ Pd-In ₂ O ₃ , ⁶⁷ Pd-TiO ₂ , ⁶⁸ Pd-V ₂ O ₅ , ⁶⁹ Pd-W ₁₈ O ₄₉ , ⁷⁰ Pd-WO ₃ , ⁷¹ Pt-SnO ₂ , ⁷² Pt-TiO ₂ ⁷³
	Other metal doping	Al-ZnO, ⁷⁴ Cd-ZnO, ⁷⁵ Co-SnO ₂ , ⁷⁶ Co-ZnO, ⁷⁷ Cr-ZnO, ⁷⁸ Eu-SnO ₂ , ⁷⁹ Mg-In ₂ O ₃ , ⁸⁰ Nb-TiO ₂ , ⁸¹ W-ZnO ⁸²
Binary metal oxide nanostructures	MOx nanoscale heterostructures	CuO-TiO ₂ , ⁴⁶ ZnO-SnO ₂ , ⁸³ Co ₃ O ₄ -SnO ₂ , ⁸⁴ CeO ₂ -In ₂ O ₃ , ⁸⁵ Nb ₂ O ₅ -CuO, ⁸⁶ Nb ₂ O ₅ -TeO ₂ ⁸⁷
	MOx-carbon nanocomposites	CNT-Co ₃ O ₄ , ⁸⁸ C-WO ₃ , ⁸⁹ graphene-In ₂ O ₃ , ⁹⁰ rGO-NiO, ⁹¹ CNF-ZnO ⁹²
Ternary or more complicated nanostructures	—	Pd-Al ₂ O ₃ -TiO ₂ , ⁹³ Pd-SnO ₂ -MoS ₂ , ⁹⁴ Pt-Pd-ZnO, ⁹⁵ rGO-Ni-ZnO, ⁹⁶ Pt/F-MWCNTs/TiO ₂ ⁹⁷

we summarize and discuss the synthetic strategy of different MOx nanomaterials with excellent H₂ sensing performance. Subsequently, we introduce MOx nanomaterials for H₂ gas sensors and emphasize methods to enhance the sensor performance by classifying mono-MOx nanomaterials, noble metal or non-noble metal-decorated MOx nanomaterials, binary semiconductor MOx nanostructures (nanoscale heterostructures, MOx-carbon nanocomposites), and ternary or more complicated nanostructures. Finally, conclusions are drawn and the future of H₂ gas sensors is prospected. Table 1 shows some major MOx nanomaterials for H₂ sensing with different number of units in structures.

2. Structure, property, and mechanism of H₂ gas sensors

Similar to other chemical sensors, gas sensors are also composed of sensing materials and signal transduction systems.⁹⁸ During the detection process, the target gas interacts with the surface of the sensing materials and changes the physical or chemical properties on the surface of the sensing material; then, the signal transduction systems convert these changes into readable signals and output.^{99,100} Therefore, the choice of sensing devices as well as the combination of materials and physical actuators should be considered for the desired sensor performance. In this section, the configurations and mechanisms of gas sensors as well as the methods of integration of sensing materials are introduced.

2.1 Sensor devices

Usually, there are two main types of prototype devices used in MOx gas sensors, which are planar and tubular devices.¹⁰¹ To meet the working environments and deformation requirements of

the equipment, flexible devices are gaining attention gradually. The advantages and disadvantages of the three configurations are shown in Table 2. Next, the structure of these three prototype devices will be explained. The first is a traditional ceramic tube device, which is composed of a ceramic tube, a signal electrode, and a heating electrode, as shown in Fig. 1a. Its core is a small ceramic tube of Al₂O₃, which greatly improves the consistency and mechanical strength of the gas sensor. Beside Al₂O₃, gold electrodes on both the sides are used for testing. The sensing material is smeared between the two gold electrodes and their surfaces, and then sintered at a high temperature. In addition, there are four other metal platinum wires welded to the gold electrode, which are used as wires welded on the base. Finally, in ceramic, a nickel-chromium alloy heating wire is inserted into the tube as the heating electrode of the device. The preparation process of the ceramic tube is simple and the cost is low; thus, it can be directly commercialized.

The second is a flat-type device, as shown in Fig. 1b; the Ag-Pd alloy electrode with excellent electrical conductivity is formed on the bottom layer of the ceramic substrate. The prepared material is coated or grown on the Ag-Pd alloy electrode, which has a zigzag shape in the middle of the substrate.¹⁰² Compared with the ceramic type, the flat type has a better performance because the material is directly coated on the surface of the electrode substrate. There are various preparation methods for the sensing material on the flat substrate, including spray coating, drop coating, electrospinning, and self-assembly. The last one is the flexible device, which uses organic polymer as the substrate, as shown in Fig. 1c.

In sensing applications, a variety of flexible substrates are selected, including polyimide (PI), polyethersulfone (PES), polycarbonate, polyvinyl naphthalate (PEN), polyester resin, and polyphenylene ethylene glycol formate (PET). As the substrate, polymers have many advantages, such as high transparency,

Table 2 Comparison of various gas sensing devices

Type	Advantages	Disadvantages
Ceramic tube	Simple preparation process, low cost, direct commercial conversion	Destruction of the material morphology and structure
Flat device	Maintain the material morphology and structure, various preparation methods for sensing materials	Poor portability
Flexible device	Flexible, light, transparent	Expensive, poor stability



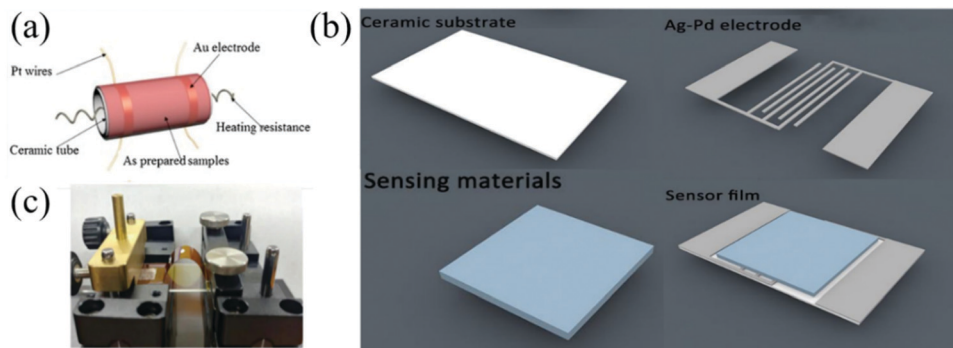


Fig. 1 Schematic diagrams of (a) the ceramic tube type gas sensor⁷⁰ (Copyright 2018, Elsevier), (b) flat gas sensor device¹⁰³ (Copyright 2020, Elsevier), and (c) polymer gas sensor device⁹³ (Copyright 2018, Wiley-VCH).

excellent flexibility, and high abrasion resistance. The polymer film can be attached on the glass and used in wearable devices due to its high transmittance and flexibility. Kim *et al.*⁹³ prepared an Al₂O₃/TiO₂ thin film heterostructure using a 2D electron gas (2DEG) on the polyimide (PI) substrate by ALD without the use of epitaxial layers or single-crystal substrates, which is promising for smart windows or other fields.

2.2 Integration of sensing materials

For MOx nanomaterials, three preparation techniques are commonly used to modify the sensor components. The first is the drop-casting process, which is extensively employed during the sensor's fabrication. In a nutshell, the sensing materials are first made into a slurry by adding some solvents such as ethanol or terpineol, and then drop-casted on the sensor substrates.^{103,104} This method is relatively simple but the phenomenon of powder shedding will reduce the life of the sensor and it is difficult to obtain the smooth sensing layers. The second method is to use various deposition methods such as atomic layer deposition (ALD), pulsed laser deposition (PLD), or chemical vapor deposition (CVD) method. These methods can deposit the target product on the surface of the substrate directly.^{95,105} For this method, the combination of the material and the substrate is relatively tight but the industrial production is limited due to its high cost. The last one is the liquid-phase chemical synthesis method, which has emerged in recent years. The sensitive materials directly grow on the sensing components and the morphology of the materials is regulated by the adjusting conditions.¹⁰⁶ For example, Alev *et al.*¹⁰⁷ obtained TiO₂ nanorods *via* the hydrothermal process, which were successfully grown on the surface of the FTO conductive glass and had a certain response to H₂. The sensor prepared by this method has good compatibility between the materials and the substrates. This method can achieve large-scale preparation and has a broad development space in the future.

2.3 Key parameters of sensors

Under normal circumstances, sensitivity, speed (response-recovery rate), selectivity, and stability are the evaluation indices of the gas sensor, which are named as “4S”.^{13,104}

According to the actual operation situation, the best working temperature and the detection limit should also be considered.

2.3.1 Sensitivity. Sensitivity is used to characterize the sensor's performance to the target gas.¹⁰⁸ For resistive gas sensors, it is usually expressed by the magnitude of resistance changes.¹⁰⁹ For different substrate materials, there are many ways to express the sensitivity of the sensors. The most common is R_a/R_g (for reducing gases such as H₂),¹¹⁰ which is often used when the resistance changes sharply after contacting with the target gas. As for the case where the resistance changes little, the sensitivity is usually expressed in $\Delta R/R$. In the actual detection process, the ratio of the magnitude of the current can also be used as a representation of the sensitivity, such as I_a/I_g ¹¹¹ and $\Delta I/I_a$.¹¹² The sensitivity of the gas sensor is closely related to the size of the effective area of the gas sensor and the number of surface active sites.

2.3.2 Response & recovery time. The response time refers to the speed of the gas sensor to the target gas at a certain operating temperature. The recovery time refers to the speed at which the gas sensor returns to its original state after the target gas is removed at a certain operating temperature.¹¹³

The response/recovery time reflects the reversibility of gas adsorption and desorption on the surface of materials. Some materials have baseline drift due to poor reversibility, which affects the performance of the sensors. The shorter response/recovery time, the better the sensor's performance.^{114,115} Rapid response is necessary for the detection of explosive gases, especially for H₂.

2.3.3 Selectivity. Selectivity refers to the specific recognition of the gas sensor and the anti-interference ability of other gases in the same environment.¹¹⁶ The selectivity is very crucial for gas sensors. Sensors with poor selectivity will exhibit an “error report” in an interfering gas atmosphere. Due to the sensing mechanism of metal oxide nanomaterials based on the surface reaction of materials, a variety of gases with similar properties will produce a response during the detection process. Therefore, improving the selectivity of the sensor by adjusting the sensing materials has become a hot topic in current research areas.^{117,118}

2.3.4 Stability. Stability is a key parameter for the industrialization and commercialization of sensors.¹¹⁹ The stability can be classified into two types. The first is the reproducibility



of the sensor during a certain period time under the working state (at the optimal working temperature in the atmosphere of the gas to be measured), which is also known as the active stability. The other refers to the ability of the sensor to maintain its original sensitivity and selectivity after storage for a long time in a relatively stable environment.¹⁰⁰ The stability of the sensor is the premise of data authenticity.

2.3.5 Working temperature. Since the movement and migration of conductive carriers in MOx nanomaterials are closely related to the environmental temperature, the sensor can work efficiently under certain temperature conditions (mostly 200–400 °C).¹²⁰ As the temperature rises, the movement and migration of conductive carriers in the material will accelerate and *vice versa*. The temperature will also influence the concentration and types of oxygen anions on the surface of the material and the thickness of the electron depleted on the layer; thus, the sensing performance will be affected. However, the temperature should not be too high because gas adsorption is an exothermic reaction and high temperature will lead to gas desorption. Furthermore, high temperature will accelerate the degradation of the sensor.¹²¹ Therefore, choosing the right working temperature is primary for gas detection.

2.3.6 Detection limit. Detection limit is defined as the lowest detection concentration that can be distinguished from noise, *i.e.*, the lowest gas concentration that the sensor can respond to.¹¹⁰ The corresponding value of the detection limit is at least 3 to 5 times that of the noise.¹²² A lower detection limit indicates that the sensor can respond to a trace concentration of gas atmosphere. At present, H₂ sensors are gradually realizing the transition from the ppm to ppb level detection.

These parameters are used as a reference for evaluating the gas sensors' performance. In actual applications, the specific parameters are adjusted depending on the environment and the purpose of the sensor.

2.4 Sensing mechanisms

Exploring the mechanism of the gas sensor helps us to get in-depth insights about the sensor and also provides sufficient theoretical support for the construction of the new gas sensing platforms in the future.¹²³ In addition, with the continuous development of computational chemistry, the working mechanism of a gas sensor is more intuitive to show in front of us by molecular simulation of the desorption process of hydrogen molecules on the surface of MOx.

At present, the mechanisms of gas sensing can be divided into two categories. From the macro-perspective, the mechanism refers to the analysis of the interaction between the material and the target gas. Ji *et al.*¹²⁴ made a detailed analysis of the macro mechanism of gas sensing. In this part, the gas sensing mechanism will be concisely introduced from the micro-perspective by the surface space charge model and the grain boundary barrier model. These two models are widely applied in the explanation of the gas sensing mechanism, irrespective of mono-MOx or MOx composites. Furthermore, some simulation studies on MOx for H₂ sensing are included.

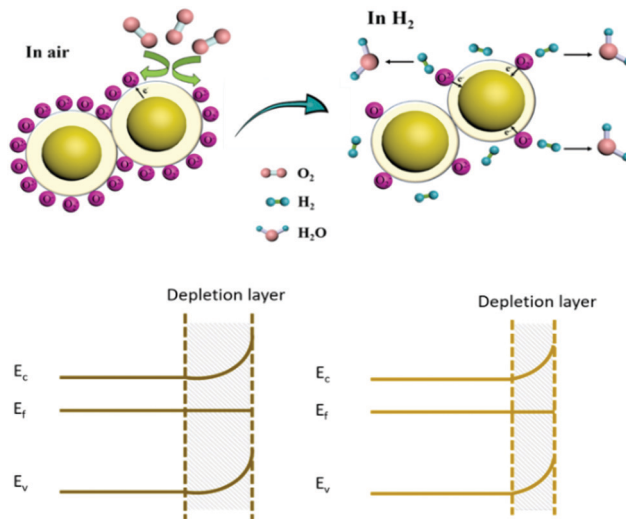
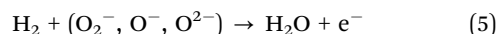
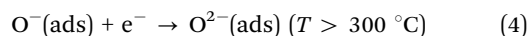
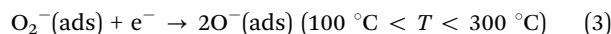
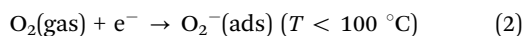
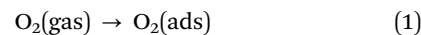


Fig. 2 Surface space-charge model.

The gas sensing mechanism of MOx nanomaterials is essentially through the adsorption and desorption of the gas on the surface of the material, and electron transfer occurs on the surface of the material, resulting in a change in the resistance.^{125,126}

Surface space-charge model: taking an n-type semiconductor as an example, the gas sensing process can be divided into two steps. As shown in Fig. 2, when the sensor is placed in an air environment, a number of oxygen molecules are adsorbed on the surface of the material and form different oxygen anions (O₂⁻, O⁻, O²⁻) as well as a space charge depletion layer, which reduce the electrons in the conduction band of the metal oxide nanomaterials and increase the resistance.⁴³ When H₂ is introduced into the chamber, it reacts with the oxygen anion and produces H₂O, resulting in the charge transfer process from the adsorbed oxygen species back to the conduction band of the MOx nanomaterials and the space charge depletion layer on the surface of the material is reduced; thus, the resistance of the sensor is reduced.¹²⁷ For p-type semiconductors, the direction of charge transfer is opposite to that of n-type semiconductors but the basic sensing mechanism is almost similar. The reactions for the formation of different oxygen anions are as follows.^{128–130}



Barrier model of grain boundary: this model is based on the theory of the semiconductor energy band, which is suitable for polycrystalline MOx.^{12,131} The main content is that the semiconductor material is a polycrystalline structure composed of many small grains and there is a grain boundary barrier



between the grains, as illustrated in Fig. 3. For n-type MO_x, when in an air atmosphere, the oxygen molecule accumulates and adsorbs on the grain boundary. Afterwards, the electron transfers between the oxygen atom and the semiconductor material. The oxygen atom gets electrons and becomes the adsorbed oxygen ion with a negative charge, which causes the potential barrier of the material surface grain boundary to rise and the electrons can no longer transition between the grains; thus, the material resistance increases.¹³² When the material is in a reducing gas such as H₂, it reacts with the adsorbed oxygen ions and electron transfer occurs again. Electrons return to the semiconductor material, the surface barrier decreases, and the material resistance decreases.¹²⁶ A change in the resistance can detect the change in the gas concentration.

The above two models are the most common gas sensing mechanism interpretation methods; however, due to the variety of gas sensing materials, there are differences in their gas sensing mechanisms. The existence of molecular simulation can make us better understand the gas sensing process of gas molecules on the surface of MO_x at the atomistic scale.^{133,134}

In gas sensing, density functional theory (DFT) calculation is the most commonly used. DFT was originally a method to deal with multi-electron systems and now it has gradually developed into a powerful tool to explore the intrinsic properties of MO_x nanomaterials and the influence of MO_x active sites on the gas adsorption and desorption process.¹³⁵ The improvement of the H₂ sensing performance caused by the defects and the selectivity of H₂ for different exposed crystal faces can be calculated by theoretical simulation.

For example, Zhang *et al.*¹³⁶ studied the gas sensing characteristics of H₂ on the TiO₂ surface with oxygen vacancy defects by DFT calculations and found the best adsorption sites of H₂ on the TiO₂ surface; also, it was further revealed by the density of states (DOS) that the presence of oxygen vacancies is beneficial for decreasing the adsorption energy of H₂ on the TiO₂ surface and for the adsorption of H₂ on the TiO₂ surface. In the same way, Umar *et al.*¹³⁵ constructed four hydrogen adsorbed systems and came to the conclusion that the oxygen anion species significantly affects the electronic structure and band gap energy of SnO₂, thus ultimately affecting the performance of the H₂ sensor.

At the same time, we can also calculate the adsorption energy of gas molecules on a certain exposed crystal face of gas sensing materials and judge the ease of gas molecule adsorption on a certain crystal face. In summary, theoretical calculations and molecular simulation are highly instructive for the study of the gas sensing mechanism of MO_x H₂.

3. Mono-semiconductor metal oxide (MO_x) nanomaterials for H₂ gas sensors

Mono-semiconductor metal oxide nanomaterials have the longest application history in H₂ gas sensors since ZnO nanostructures were first applied for flammable gases in 1962. Both the morphology and structure of different materials can influence the performance of the sensor. Mono-MO_x nanomaterials are favored by researchers due to the facile and inexpensive synthetic methods as well as the aesthetically pleasing nano-/micro-structure. In this section, both the fabrication strategies and the application in H₂ sensing are detailed.

3.1 Structure effect

As mentioned above, the basic mechanism of gas sensors relies on the change in the sensor resistance by the reaction between the adsorbed target gas molecules and the chemisorbed oxygen species on the sensing material surface; thus, the gas sensing ability of mono-MO_x nanomaterials is closely related to the size, structure, and morphology. The nano-/micro-structure with outstanding porosity and tunable large surface–volume ratio could overwhelmingly enhance the performance of sensors.

Different morphologies of MO_x nanomaterials mean different BET surface areas. High BET surface area is beneficial for increasing the contact between the gas and the material surface so as to improve the gas sensing performance.^{25,137}

It was calculated from the free dimension in the spatial dimension that was not constrained by the nanoscale that the MO_x nanomaterial can be classified as a one-dimensional (1D) nanostructure, two-dimensional (2D) nanostructure, or three-dimensional (3D) nanostructure.⁹⁹ The unique advantages of each structures are illustrated in Table 3.

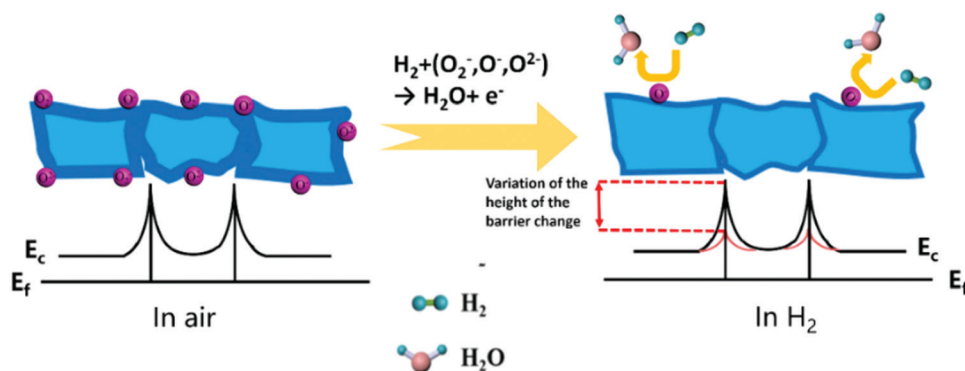


Fig. 3 Barrier model of grain boundary.



Table 3 Comparison of MOx in different dimensions

Dimension	Representative structures	Advantages
1D	Nanofibers, nanowires, nanotubes	High BET surface area, high density of reactive sites, high length–width ratio
2D	Nanofilms, nanosheets	Large surface to volume ratio, fast electron transfer rates
3D	Nanourchins, nanoflowers, nanoclusters	High specific surface areas, fast and effective gas diffusion

In addition, the porosity also has a crucial influence on the gas sensing performance.¹³⁸ The introduction of a porous structure into the MOx will increase the porosity of the material and the gas sensing properties of the material can also be significantly improved because the pore size and pore size distribution will affect the gas diffusion process on the surface of gas-sensing materials.¹³⁹ Among the various porous structures, hierarchical porous structures have their unique advantages. Firstly, the hierarchical structure provided a large number of surface reactive active sites, which effectively promoted the reaction of gases on the surface of sensitive materials and greatly improved the sensitivity of the sensors; moreover, the porous structure offers many conveniences for the transport of electrons and the diffusion of gases inside the pore channels, which shortens the response time of the materials to gases.

3.2 Synthetic strategy

High-performance sensing materials can be obtained by controlling the morphology and structure of the MOx nanomaterials,^{50,56,140} especially in the atomic and molecular dimensions. The synthetic methods have a great impact on the morphology of the materials.^{141–143}

Some common preparation methods will be introduced in this section, such as magnetron sputtering technology, CVD, thermal evaporation method, sol–gel method, electrospinning technology, and hydrothermal solvothermal method. The details of the synthesis are listed in Table 4.

Magnetron sputtering, as a common physical vapor deposition technology, is one of the important methods for the preparation of high-quality MOx films.¹⁵⁹ It can obtain high purity thin films by the direct interaction of one or more target materials with the reaction gas and the coating has better bonding strength with the substrate.

In the preparation process, the type of the target material, as well as the composition and proportion of the carrier gases can be adjusted to obtain a high-quality film. In the choice of the target, the pure metal target or MOx target can be utilized according to the need. Meanwhile, pure Ar or the mixture of Ar and O₂ can be introduced into the chamber in the sputtering process. The distance between the target and the substrate, sputtering pressure, sputtering time, annealing temperature, sputtering power, and other conditions can directly influence the structure and properties of the formed films.¹⁶⁰ For instance, Abubakar *et al.*¹⁴⁹ prepared nanostructured NiO on ITO conductive glass by a facile but novel sputtering oxidation coupling method. High-quality NiO thin film was obtained by oxidizing the Ni thin films by a hot-dry treatment. Rahmani *et al.*⁴⁰ also synthesized WO₃ nanosheets by the same technology.

Magnetron sputtering technology has the advantages of a simple preparation process and low cost but the substrate or film might be damaged as it is a high-energy process.

CVD is another kind of technology in which the precursors react with each other in a gas atmosphere and then the materials are deposited on the surface of the heated solid matrix. CVD has been widely used to prepare MOx nanomaterials.^{161,162} The CVD device is shown in Fig. 4e. In the reaction process, the morphology and composition of the films can be controlled by regulating the gas flow rate and the reaction temperature.

Pradeep *et al.*¹⁵⁵ synthesized a series of MgO nanocubes by CVD under different temperatures (800 °C, 900 °C, 1000 °C, 1100 °C); the SEM images are shown in Fig. 4a–d. With the increase in the reaction temperature, the morphology of MgO cubes gradually becomes uniform. In order to accommodate more needs, some new technologies for CVD are emerging. For example, Stoycheva *et al.*¹⁵⁶ obtained WO₃ nanoneedles using aerosol-assisted chemical vapor deposition technology (AACVD). The TiO₂ thin films can be fabricated by metal–organic chemical vapor deposition (MOCVD).⁴³

In addition, the thermal evaporation method is used to vaporize the precursors in noble gases (or reactive gases) and then form a thin film by collision, cooling, and condensation processes. According to the precursors used in the preparation process, the thermal evaporation method can be generally divided into the direct evaporation of MOx powder, pure metal powder thermal evaporation, and decomposition of MOx. Different types of high-quality MOx nanomaterials can be produced simply by thermal evaporation through a cheap deposition system. The schematic diagram of its deposition equipment is shown in Fig. 4f. In the preparation process of sensitive materials for H₂ sensors, pure metal powder thermal evaporation is mostly used.

For instance, CuO nanocrystals were grown on copper foil by thermal oxidation at 600 °C.¹¹¹ Similar to other vapor deposition technologies, the morphology and structure of products resulting from thermal evaporation are affected by the working parameters such as temperature, carrier gas, evaporation source, catalyst, and pressure. San *et al.*¹⁵⁸ prepared ZnO nanorods with different morphologies by controlling the evaporation temperature and growth time. The effect of temperature on the structure of ZnO nanorods is shown in Fig. 4g–i. At the evaporation temperature of 900 °C, the surface of ZnO nanorods is smooth and uniformly distributed. When the evaporation temperature rises to 975 °C, hexagonal nanorods with tips are formed, and very short nanorods with diameters of 50 nm and lengths of 100 nm are observed at the top of each ZnO nanorod. When the temperature was further increased to 1050 °C, the morphology of ZnO changed again and a hierarchical nanorod structure with a decreasing diameter from top to bottom was obtained. The effect of the growth time is shown in Fig. 4j–l. It can be observed that the morphology of ZnO gradually improved

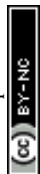


Table 4 Details of mono-semiconductor metal oxide nanomaterial synthesis

Materials	Method	Morphology	Size	Temp, time	Conv. cond.	Ref.
ZnO	Electrospinning	Nanofibers	d : 120 nm	15 kV, 0.07 mL h ⁻¹	600 °C, 2 h	12
SnO ₂	Electrospinning	Porous hollow nanofibers	L : 0.5–2 μm d : 120 nm	15 kV, 0.4 mL h ⁻¹	600 °C, 3 h	127
ZnO	Electrospinning	Nanofibers	Nanograins: 30 nm	15 kV, 20 cm, 0.05 mL h ⁻¹	600 °C, 0.5 h	131
ZnO	Hydrothermal	Dumbbell-shaped	L : 1–1.4 μm, ends: 300–400 nm, mid-section: 200–300 nm	160 °C, 10 h	—	41
VO ₂ (A)	Hydrothermal	Nanobelts	L : ≥ 20 nm, δ : less than 10 nm	230 °C, 48 h	—	144
In ₂ O ₃	Hydrothermal	Mesoporous nanoparticles	5–20 nm	150 °C, 10 h	500 °C, 2 h	114
MoO ₃	Hydrothermal	Nanoribbon	δ : 90 nm, L : 20 nm, W : 270 nm	220 °C, 12 h	300 °C, 2 h	132
TiO ₂	Hydrothermal	Nanoflower	2–3 μm	170 °C	—	136
WO ₃	Hydrothermal	Urchin-like	0.5–2 μm	180 °C, 24 h	—	141
MoO ₃	Hydrothermal	Nanoribbons	W : 400 nm, L : 500 μm	260 °C, 24 h	—	145
SnO ₂	Hydrothermal	2D disks	δ : 1 μm	160 °C, 12 h	450 °C, 2 h	135
Nb ₂ O ₅	Hydrothermal	Nanorod arrays	W : 234 ± 20 nm, L : 702 ± 128 nm	175 °C, 15 h	—	146
α-MoO ₃	Hydrothermal	Nanoribbon	W : 200 nm	200 °C, 12 h	—	147
Bi ₂ O ₃	Microwave irradiation method	Grape-like Worm-like	Rugged spheres d : 1 μm, L : 5 μm	180 °C, 24 h 180 W, 15–20 strokes	300 °C, 1 h	148
SnO ₂	DC sputtering	Nanofilms	—	25 °C	450 °C, 1 h	143
WO ₃	RF sputtering	Nanosheets	W : 50–500 nm, δ : 10–50 nm	300 °C, 20 min	450 °C, 5 h	40
ZnO	RF sputtering	Nanotubes	δ : 42.7 nm	40 min	—	130
NiO	RF sputtering	Thin film	W : 80–200 nm	150 °C, 60 min	550 °C, 2 h	149
NiO	Sol-gel method	Thin film	δ : 21 nm	25 °C	550 °C, 3 h	150
ZnO	Sol-gel method	Thin film	Grain size: 75 nm	RT, 48 h	350 °C	151
SnO ₂	Sol-gel method	Thin film	δ : 595 nm	70 °C, 8 h; RT, 24 h	500 °C, 2 h	152
SnO ₂	Sol-gel method	Thin film	Crystallite size: 33.19/33.20 nm	70 °C, 8/10 h	400/500 °C, 2 h	153
NiO	Sol-gel method	Thin film	Porosity: 24% Grain size: 27 nm Porosity: 35% Grain size: 17 nm	25 °C, 2 h 25 °C, 2 h	550 °C, 3 h one time 550 °C, 3 h several times	154
MgO	CVD	Nanocubes	100–200 nm	800/900/1000/ 1100 °C	—	155
WO ₃	AACVD	Nanoneedles	Grain size: 50–100 nm	350/450/500 °C	—	156
TiO ₂	MOCVD	Thin films	δ : 71/103/381 nm	500 °C	—	43
TeO ₂	Thermal evaporation method.	Nanowires	d : 30–40 nm, L : 20 μm	330 °C, 1 h; distance: 2 mm	—	61
CuO	Thermal evaporation method.	Nanowires	d : 120 nm	600 °C, 6 h	—	111
SnO ₂	Thermal evaporation method.	Nanorods	d : 30 nm, L : several hundred nanometers	900 °C, 1 h	—	157
ZnO	Thermal evaporation method.	Nanorods	d : 50–120 nm, L : 1–6 μm	900/975/1050 °C, 5/ 10 min	—	158
SnO ₂	Thermal evaporation method	Nanorods Nanowires	d : 30 nm, L : several tens to several hundreds of nanometers d : 30–200 nm, L : several tens of micrometers	900 °C, 1 h 900 °C, 1 h	O ₂ + Ar Ar	143

with the growth time. It can be seen that the structure and morphology of MOx nanomaterials can be adjusted by controlling the evaporation temperature and growth time.

Moreover, Shen *et al.*¹⁴³ achieved the synthesis of SnO₂ nanorods to nanowires by using different types of carrier gas and changing the location relationship between the evaporated powder and the substrate. Because the environment of the material prepared by the thermal evaporation method is relatively closed and the carrier gas is only Ar or O₂, the prepared MOx nanomaterials are highly pure. Moreover, the method is easy to operate, the system control of the instrument and the equipment are simple, and the synthetic cost is relatively low. However, the disadvantages of this method is also obvious due to the high temperature and poor repeatability in the preparation process, making it difficult to achieve controllable growth.

The sol-gel method could synthesize materials at low temperature or even at room temperature,¹⁶³ which is widely used

to prepare bulk materials, powder materials, fiber materials, films, and coatings.¹⁶⁴ The process of preparing semiconductor MOx films by the sol-gel method is relatively simple.¹⁶⁵ Due to the three-dimensional grid structure generated during the aging process, the prepared film with a high specific surface area is obtained,^{76,166} which promotes the efficient and sensitive detection of sensing materials for H₂.

The precursors of the sol-gel method are roughly divided into two kinds, metal alkoxides and metal inorganic salts. The former are relatively mature in controlling the reaction process but their costs are relatively high. In addition, the extremely high activity limits their large-scale use. Conversely, the metal inorganic salts are inexpensive and easy to industrialize.

In the process of sol-gel preparation, the hydrolysis and polycondensation reaction of metal alkoxide(s) are the key to the success of preparation; thus, the molar ratio of water to alkoxide is crucial in the preparation process. Meanwhile, the



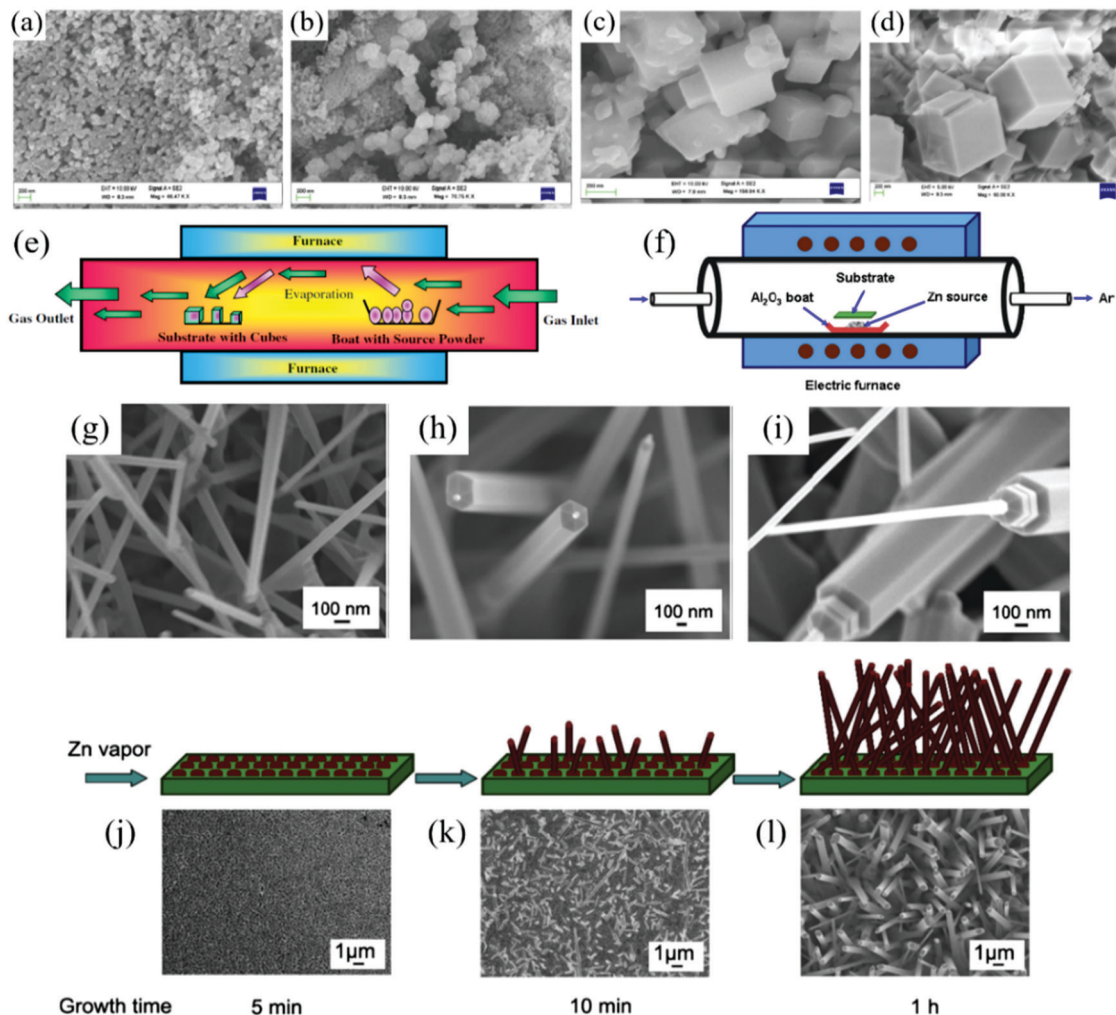


Fig. 4 SEM images of nanocubes grown at different temperatures (a) 800 °C; (b) 900 °C; (c) 1000 °C; (d) 1100 °C; (e) conventional approach for the growth of nanocubes using CVD system¹⁵⁵ (Copyright 2017, Springer Nature). (f) Schematic diagram of the apparatus used for the preparation of ZnO nanostructures; FESEM images of ZnO nanostructures synthesized at different source temperatures: (g) 900 °C; (h) 975 °C; (i) 1050 °C; schematic illustration of the growth process of ZnO nanostructures. The SEM image is for the product synthesized at 975 °C during different growth times: (j) 5 min; (k) 10 min; (l) 1 h¹⁵⁸ (Copyright 2015, Elsevier).

thickness of the film can be changed by changing the spin coating speed.¹⁵² In the last step of film formation, the pore size and pore volume of the film can be controlled by controlling the calcination temperature, realizing the regulation of the morphology and structure of the film. Compared with other film preparation methods (CVD, PVD), the biggest advantage of the sol-gel method is the low temperature of the preparation process. Moreover, the sol-gel method belongs to solution-phase synthesis; thus, it can obtain some unique structures easily. A series of metal oxide nanomaterials have been prepared for gas sensing by this method, such as NiO¹⁵⁰ and ZnO.¹⁵¹

Electrospinning technology is considered as a significant method to prepare ultrathin nanofibers.¹⁶⁷ From the basic principle, electrospinning technology can be seen as a special form of electrostatic atomization or electrostatic spraying. In the electrospinning process, the atomized ejected material is not a tiny droplet but a charged liquid flow. In this process,

liquid flow is greatly stretched under the high voltage electric field, thus forming micro/nanofibers.^{168,169}

In early 2007, researchers prepared the first SnO₂ nanofiber gas sensor *via* the electrospinning technology. The sensor has a sensitive response to water vapor and formaldehyde.¹⁷⁰ After continuous development, the electrospinning technology has been widely used to prepare morphologically and compositionally controllable MOx nanomaterials.¹⁷¹ Kim *et al.*¹² prepared ZnO nanofibers with a diameter of approximately 120 nm by the electrospinning technique. The specific preparation process is shown in Fig. 5a. Firstly, 10% polyvinyl acetate (PVA) was mixed with zinc acetate solution to obtain the electrospinning solution and then the mixed solution was loaded into the electrospinning nozzle. During the electrospinning process, a positive voltage (+15 kV) and negative voltage (−10 kV) was applied to the needle and aluminum collector, respectively. Electrospinning was carried out at room temperature with a feed rate of the solution of 0.07 mL h^{−1} and the calcination condition of 600 °C for 2 h to



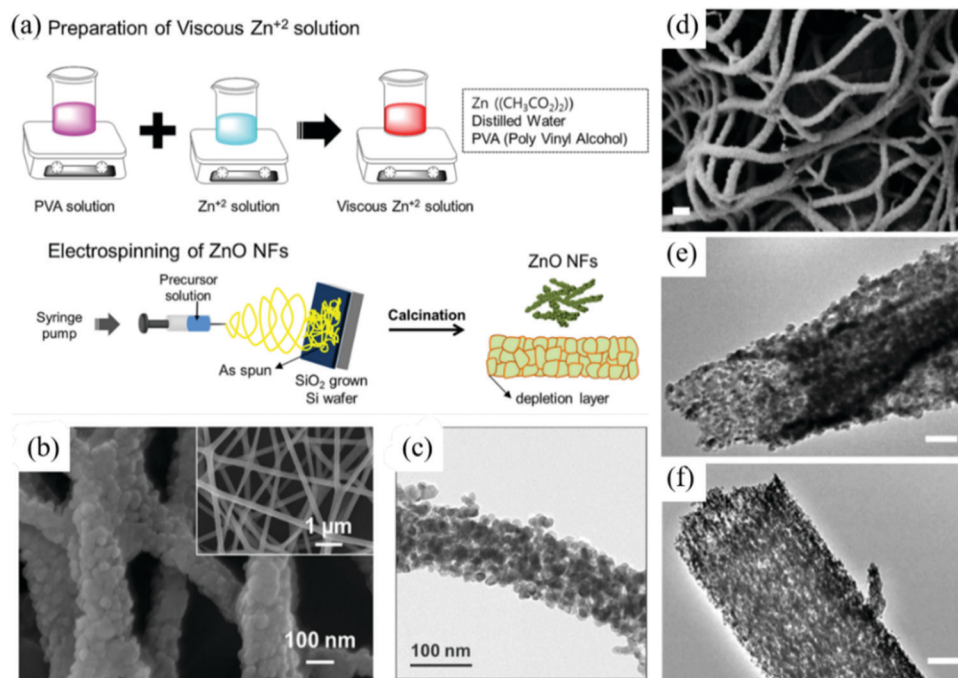


Fig. 5 Schematics for (a) the preparation of ZnO NFs by the electrospinning process; (b) the FE-SEM image of ZnO; (c) TEM image of ZnO NFs¹² (Copyright 2019, Elsevier). (d) SEM images of the SnO₂ nanofibers; (e), (f) TEM images of the SnO₂ hollow nanofibers¹⁷² (Copyright 2014, American Chemical Society).

remove the solvent and PVA in order to obtain the ZnO nanofibers. The morphology of the films is shown in Fig. 5b and c.

As a newly developed method of nanofibers, electrospinning has unique advantages, especially for the preparation of nanofibers with high porosity. Nanofibers prepared by the electrospinning technology have good interconnection and relatively uniform pore size distribution. Hollow SnO₂ nanofibers have been synthesized by a single nozzle electrospinning process using phase-separated mixed polymer,¹⁷² as shown in Fig. 5d–f. The high specific surface area is beneficial for the adsorption of gases and the formation of defects, which could exert a positive impact on improving the performance of the sensors in all the aspects. However, it is difficult to obtain nanofibers with fixed orientation *via* electrospinning.¹⁷³

There is no denying that hydrothermal or solvothermal methods play an extremely important role in the synthesis of MOx nanomaterials.¹⁷⁴ Compared with traditional materials preparation methods, the hydrothermal method can greatly control the nucleation process and the crystallinity of the materials. Recently, the study of hydrothermal synthesis through thermodynamic calculation provides an incentive for material preparation development. A series of MOx nanomaterials with different morphologies, such as nanorods,¹⁴⁶ nanoribbons,^{132,147} nanoflowers,¹³⁶ and nanoporous structures,¹¹⁴ are obtained by the hydrothermal or solvothermal methods.

The essence of the hydrothermal method is the recrystallization process, during which the nucleation and growth of grains occurs in the autoclave. Zhang *et al.*¹⁴¹ synthesized sea urchin-like hexagonal WO₃ nanostructure *via* the hydrothermal method. The growth process of WO₃ is shown in Fig. 6a. The capping

agent K₂SO₄ is added during the synthetic process to promote the anisotropic growth of WO₃, resulting in a sea urchin-like hexagonal WO₃ nanostructure with a large specific surface area. The SEM images of WO₃ are shown in Fig. 6b–d.

The hydrothermal or solvothermal methods for MOx nanomaterial synthesis are usually affected by a few factors such as the reaction temperature, reaction time, the concentration of reactants, and the ratio of structure-directing agents.¹⁷⁵ First, the temperature can influence the activity of the substances in the chemical reaction process, the types of products, as well as the grain size of the products. Then, the crystallinity gradually increases with the extension of the hydrothermal reaction time. Last but not the least, the change in the reaction medium can not only affect the solubility of the precursors and the growth rate of crystal but also, more importantly, change the structure of the growth unit in the solution, and ultimately determine the structure and shape of the materials. Therefore, the choice of appropriate conditions has a profound influence on the morphology of materials and hence affects the H₂ sensing performance ultimately.

The hydrothermal method is usually combined with the annealing process, which can not only convert some hydrothermal precursors into MOx but also generate more oxygen vacancies on the surface of MOx nanomaterials.¹⁷⁶ However, the two-step preparation has a risk of contaminating the samples and the energy consumption of high-temperature annealing.

In this regard, further development of the hydrothermal method is the “one-pot” method, known as the one-step direct synthesis of MOx nanomaterials. By controlling the reaction



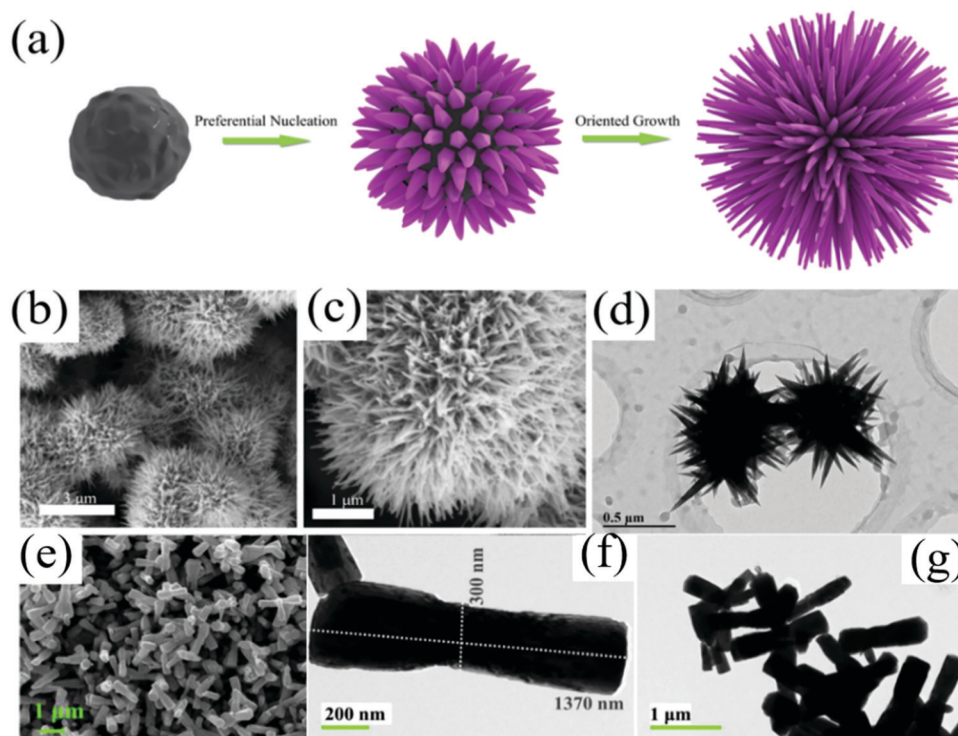


Fig. 6 (a) Schematic diagram of the formation mechanism of sea-urchin-like WO_3 ; (b and c) SEM micrographs of sea-urchin-like WO_3 with different magnification; (d) TEM images of the sea-urchin-like WO_3 ¹⁴¹ (Copyright 2019, Elsevier). (e) FESEM micrograph of dumbbell-shaped ZnO ; (f and g) TEM images of dumbbell-shaped ZnO ⁴¹ (Copyright 2020, Elsevier).

conditions, various materials with special morphologies can be obtained. Kumar *et al.*⁴¹ adjusted the pH of the precursor solution to 10 by adding NH_4OH and obtained dumbbell-like ZnO structures. The morphology is shown in Fig. 6e–g. SnO_2 ,¹³⁵ MoO_3 ,¹⁴⁷ Nb_2O_5 ,¹⁴⁶ and other MOx nanomaterials with good morphology were prepared by the “one-pot method”. The MOx nanomaterials synthesized by the “one-pot method” are simple, low-cost, and eco-friendly. Moreover, they could avoid the impurities and structural defects introduced by high-temperature calcination. However, it can be easily found from Table 3 that the MOx nanomaterials prepared by the “one-pot method” have a higher reaction temperature and longer reaction time, which puts forward higher requirements for autoclaves.

Despite the many obvious advantages of the hydrothermal or solvothermal methods, the mechanism of the relationship between the morphology, interfacial tuning, and the reaction conditions are still obscure. In the future, the scale-up experiment of the hydrothermal method in industrial production will require continuous development scientifically and technologically.

In order to overcome some of the disadvantages of hydrothermal synthesis, several new techniques, for example, microwave-assisted synthesis have been developed in recent years. Microwave heating is internal heating, which has the characteristics of high heating speed, uniform heating without a temperature gradient, and no hysteresis effect, which can reduce the reaction temperature and shorten the reaction time.¹⁷⁷ Shinde *et al.*¹⁴⁸ obtained worm-like bismuth oxide nanostructures by the microwave-assisted method.

3.3 N-Type semiconductor metal oxide nanomaterials

N-Type semiconductor metal oxides are electronically conductive and have unique advantages in the field of gas sensing, thus becoming the most widely used in mono-metal oxide nanomaterials. Table 5 summarizes the details of some n-type semiconductor MOx nanomaterials used for H_2 sensors.

Among them, ZnO nanostructures with wide bandgap ($E_g = 3.3$ eV at RT) and high exciton binding energy (60 meV) are favored by researchers due to the diverse preparation methods and the controllable morphologies.^{187,188} Moreover, at a high temperature, the surface of ZnO particles will be reduced to metallic Zn during the detection of H_2 , resulting in a sharp decrease in the resistance. This semiconductor-to-metal transition (Fig. 7) enhances the macroscopic resistance modulation ability of ZnO and greatly improves the sensitivity of the sensor.¹³¹

The morphology and structure of the materials have an important impact on the performance. By virtue of the exceedingly rapid electronic transport characteristics, one-dimensional nanostructures have shown great advantages in the field of H_2 sensing. Sinha *et al.*¹¹² prepared high-quality single-crystal ZnO nanowires with diameters of 30–110 nm by the sol-gel method, which can reversibly realize the efficient detection of 100 ppm H_2 at the working temperature of 250 °C. Similarly, ZnO nanowires prepared by the catalyst-free thermal evaporation method can also realize the detection of H_2 .¹⁵⁸

In addition, high BET-surface areas usually lead to excellent sensing performance. The nanotube structure is demonstrated



Table 5 The response of n-type semiconductor metal oxide nanomaterials to H₂

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
Bi ₂ O ₃	Hierarchical worm	100 ppm	27	10 ppm	50% ^d	42/83 s	148
In ₂ O ₃	Octahedra	4 ppm	200	4 ppm	14 ^a	—	15
In ₂ O ₃	Mesoporous	500 ppm	260	0.01 ppm	18.0 ^a	1.7/1.5 s	114
In ₂ O ₃	Nanocubes	5 ppm	150	0.1 ppm	25 ^a	67/143 s	178
α-MoO ₃	Nanoflakes	1 vol%	200	0.06 vol%	58% ^a	7 s	55
MoO ₃	Nanoribbons	1000 ppm	RT	1 ppm	17.3 ^b	10.9 s	132
MoO ₃	Nanoribbon	2000 ppm	RT	—	11.2 ^c	15/13.5 s	145
MoO ₃	Nanoribbons	1000 ppm	RT	0.5 ppm	90% ^d	14 s	147
Nb ₂ O ₅	Nanorod arrays	6000 ppm	RT	1000 ppm	74.3% ^d	28 s	146
SnO ₂	Nanowires	1000 ppm	150	100 ppm	6.5 ^e	—	143
SnO ₂	Disk-like	100 ppm	400	—	14.7 ^b	4/331 s	135
SnO ₂	Nano tetragonal	1000 ppm	RT	150 ppm	2570% ^f	192/95 s	152
SnO ₂	Nanorods	3000 ppm	200	5.46 ppm	6.54 ^a	—	157
SnO ₂	Nanosheet	500 ppm	300	—	9.3 ^a	4/42 s	179
TiO ₂	Ordered mesoporous	1000 ppm	RT	100 ppm	298 ^a	85/198 s	51
TiO ₂	Thin films	1 ppm	RT	1 ppm	4% ^e	—	57
TiO ₂	Nanotubes	1000 ppm	200	250 ppm	200% ^f	—	107
TiO ₂	Nanoflower	500 ppm	400	—	26 ^a	10/13 s	136
TiO ₂	Nanorod array film	1 ppm	25	1 ppm	18% ^d	2/40 s	180
TiO ₂	Nanowires	500 ppm	400	90 ppm	5.2 ^a	—	181
V ₂ O ₅	Hollow structure	200 ppm	25	10 ppm	~2.9 ^h	30/5 s	182
WO ₃	Nanosheets	1 vol%	250	0.06 vol%	80% ^d	—	40
WO ₃	Sea-urchin-like	10 ppm	250	10 ppm	~4 ^a	—	141
ZnO	Hollow particles	100 ppm	225	2 ppm	9.15 ^a	139/2587 s	140
ZnO	Nanofibers	10 ppm	350	0.1 ppm	150 ^a	—	12
ZnO	Thick films	300 ppm	250	—	44 ^f	9 s	58
ZnO	Nanowire	100 ppm	250	10 ppm	98% ^f	60/14 s	112
ZnO	Nanowires	1000 ppm	200	—	5.3 ^a	—	158
ZnO	Porous nanotube	1000 ppm	250	10 ppm	139.11 ^a	—	130
ZnO	Nanofibers	10 ppm	350	0.1 ppm	109.1 ^a	—	131
ZnO	Nanorods	100 ppm	250	5 ppm	5.03 ^a	—	176
ZnO	Tetrapods	100 ppm	400	—	1.6 ^c	—	183
ZnO	Thin film	1200 ppm	400	—	23 ^d	110 s	184
ZnO	Nano lily-buds	80 ppm	180	—	1.78% ^e	1–2 s	185
ZnO	Holey nanosheets	100 ppm	RT	—	115% ^e	9/6 s	186

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .

to be an excellent sensing material due to the hollow structure. It has two effective aspects, resulting in the larger specific surface area than that of the typical one-dimensional nanomaterial; thus, the performance of the sensor can be predominantly improved. Park *et al.*¹³⁰ used ZnO nanotubes as the sensing material and greatly improved the performance. It is because the porous structures provide a large surface area, which leads to more absorption sites on the surface.

Some special morphologies with high surface areas can also improve the sensor response. Kumar *et al.*¹⁸⁵ prepared nano lily-buds (NL-buds) garden-like ZnO nanostructure by a simple single-step thermal decomposition method, as shown in Fig. 8a. Three-dimensional structure enhances the contact between the gas and the sensing materials, which contributes to the improvement of the sensor performance, excellent stability, and selectivity (Fig. 8b and c).

SnO₂, as another wide-bandgap n-type semiconductor MOX nanomaterial ($E_g = 3.6$ eV at RT), has also been widely used in H₂ gas sensing. Umar *et al.*¹³⁵ prepared discoidal SnO₂ by the hydrothermal method, which has a good response to the low concentration of H₂ but the working temperature of the sensor is higher than 400 °C. Due to the higher working temperature, the adsorption of H₂ on the surface of the material is relatively

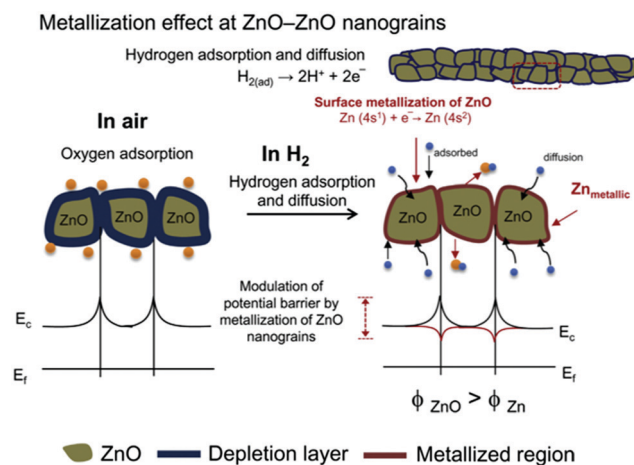


Fig. 7 The phenomenon of the semiconductor-to-metal transition of ZnO¹³¹ (Copyright 2015, Elsevier).

tight and the desorption process is slow; thus, the recovery time of the sensor is longer than 331 s. Afterwards, Choi *et al.*¹⁷⁹ used SnO₂ nanosheets as the sensing material to realize the decrease in the working temperature; this is because the prepared SnO₂ nanosheets are interconnected and these junctions



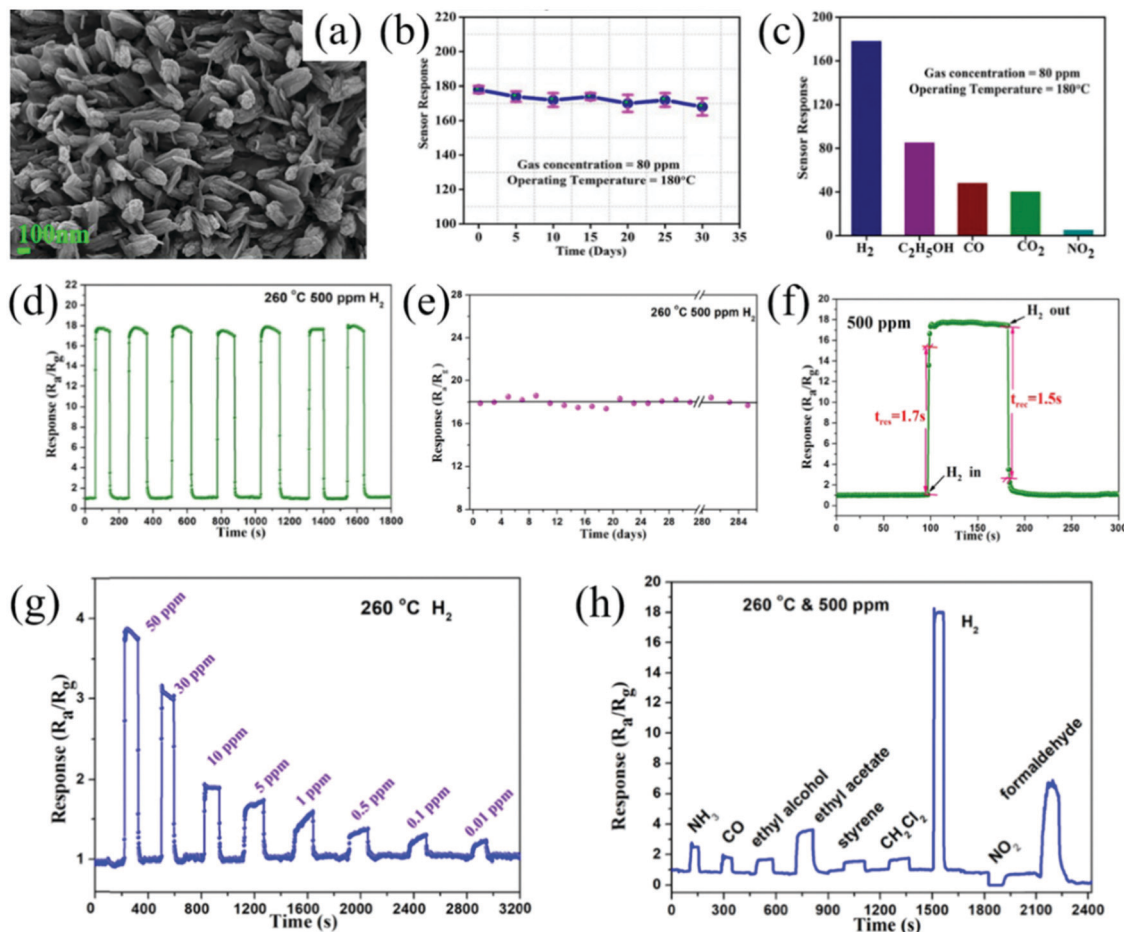


Fig. 8 (a) FESEM images of NL-buds ZnO; (b) stability of sensor; (c) selectivity at 180 °C for 80 ppm gas concentration¹⁸⁵ (Copyright 2019, Elsevier). (d) Reproducibility and (e) stability of the sensor based on mesoporous In_2O_3 for 500 ppm of H_2 at 260 °C; (f) response/recovery curves of the mesoporous In_2O_3 -based sensor to different gases with a concentration of 500 ppm at 260 °C¹¹⁴ (Copyright 2018, Elsevier).

as the channel for electron transfer accelerate the reaction between the SnO_2 nanosheets and H_2 , and enhance the sensor's response capacity to H_2 . At the same time, these channels are also beneficial for the reversible adsorption and desorption of H_2 on the material surface, which shortens the response recovery time of the sensor to 4/42 s.

Although SnO_2 and ZnO have been widely used in H_2 gas sensing, other N-type semiconductor MOx such as Bi_2O_3 , In_2O_3 , MoO_3 , TiO_2 , V_2O_5 , and WO_3 have come into researchers' attention for achieving detection at low power and high selectivity. Researchers have made achievements in the regulation of novel architectures and surface functionalization of these nanomaterials. Li *et al.*¹¹⁴ synthesized mesoporous In_2O_3 by the hydrothermal method with a high surface area. The mesoporous structure and large specific surface area are favorable for the absorption/desorption of H_2 . Therefore, the reproducibility and stability of the sensor are excellent (Fig. 8d and e), and the response/recovery speed is very fast, which is only 1.7/1.5 s. As shown in Fig. 8f, at the optimal operating temperature (260 °C), the sensor could realize detection at the ppb level (Fig. 8g). The selectivity of the sensor is also excellent as among several

interfering gases, the sensor has the highest response to H_2 , as shown in Fig. 8h.

Haidry *et al.*⁵¹ synthesized the ordered mesoporous TiO_2 nanostructure by the evaporation-induced assembly method. The sensing material can detect H_2 in a wide range from 100 ppm to 1000 ppm at room temperature and has a good linear relationship. The sensor shows a high response to 1000 ppm of H_2 in the environment. It has been reported that MoO_3 nanobelts synthesized by Yang *et al.*¹³² and Nb_2O_5 nanoarrays synthesized by Zou *et al.*¹⁴⁶ can also achieve the efficient detection of H_2 gas at room temperature.

Some post-processing methods could lower the barrier for the practical application of conventional mono-MOx nanomaterials; high-energy radiation can greatly improve the performance of the sensors. Herein, ZnO is taken as an example to introduce the application of high energy radiation in the H_2 gas sensor.

Kim *et al.*¹² placed ZnO nanofibers under the ion beams of different illuminations for irradiation and investigated the effect of electron beam irradiation on their sensing performance. As shown in Fig. 9a–d, the morphology of the materials changed under different radiation intensities. Through the irradiation of



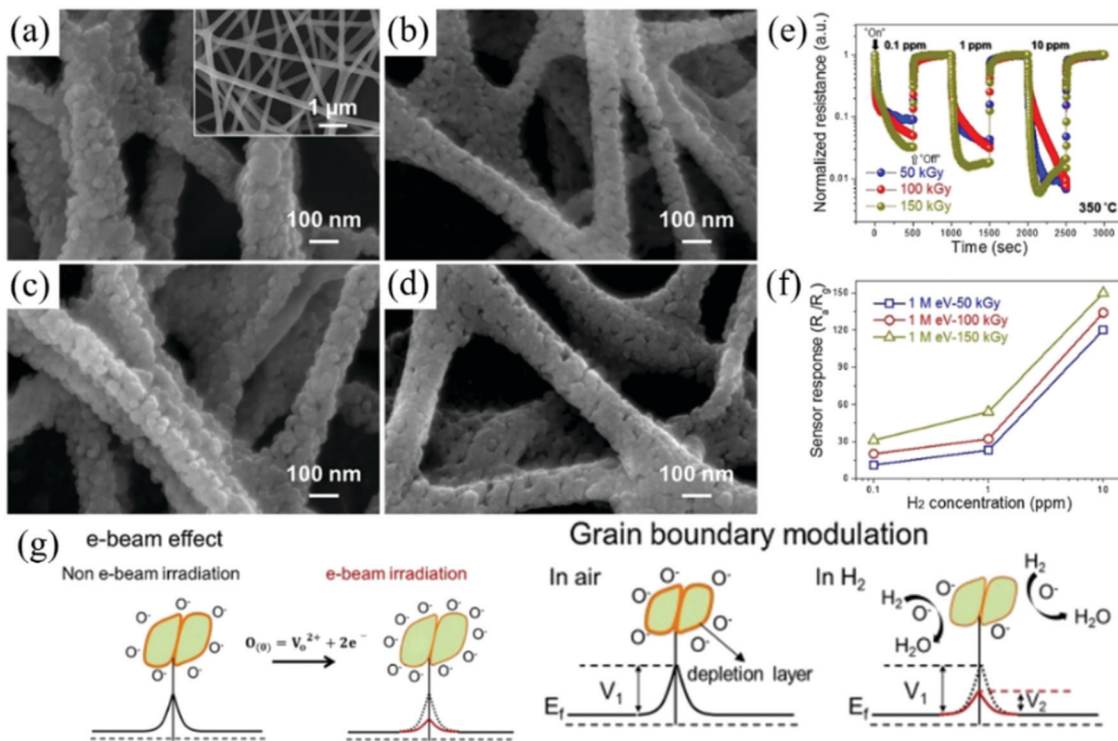


Fig. 9 FE-SEM images of ZnO NFs irradiated at doses of (a) 0 kGy; (b) 50 kGy; (c) 100 kGy; and (d) 150 kGy; (e) normalized dynamic resistances of ZnO NF sensors irradiated at different e-beam doses to 0.1, 1, and 10 ppm H₂ at 350 °C; (f) corresponding calibration curves; (g) sensing mechanism of the e-beam-irradiated ZnO NFs gas sensor¹² (Copyright 2019, Elsevier).

high-energy electron beam, the physical and chemical characteristics of the sensing material change, which provide more adsorption sites for the oxygen anion on the surface of the material (Fig. 9g). The formed oxygen vacancies are of great significance in enhancing the performance of the sensors. As seen from Fig. 9e and f, the response of the gas sensor was improved greatly.

After electron beam irradiation, the structure of the material could be changed and different types of defects could be formed, such as interstitial atoms and vacancies.¹⁵⁰ Vacancies play a key role in the electronic properties of sensitive materials and the increase in the vacancies can enhance the concentration of adsorbed oxygen on the surface, thereby improving its sensing performance.

The advantage of using high-energy radiation to modify MOx nanomaterials is that not only does it change the macro-morphology of the original material but also modulates the structural defects on the surface of the material due to the strong penetration and decomposition efficiency. It affects the excitation of the hole–electron pairs in semiconductor MOx, increasing the width of the accumulating layer and decreasing the width of the depleting layer, thus providing more adsorption sites on the surface of the material to improve the sensor performance and selectivity. These principles are also applicable to other MOx nanomaterials, which provide a new idea for the construction of a new sensing platform in the future.

Remarkably, the exposure position and area of the crystal planes also influence the performance of the sensor.^{189,190} Different crystal

surfaces have different surface energies;¹⁹¹ thus, the energy required for gas adsorption on its surface also varies. Since different exposure positions and areas affect the number of oxygen vacancies in MOx, the more the oxygen vacancies, the stronger the macroscopic resistance modulation ability of the material, and the performance of the sensor will be enhanced.

Wu *et al.*¹⁷⁶ prepared three kinds of ZnO nanostructures and found that with different exposure surfaces, the sensors exhibited different performances. Among them, the tower-like structure ZnO has the highest response to H₂, which is due to the fact that the (0001) crystal plane of the tower-like structure ZnO is exposed the most. The (0001) crystal plane contains more oxygen vacancies than the other planes, which means that on the (0001) crystal plane, the number of adsorbed oxygens will be greatly increased, which is conducive for the improvement of the sensor performance. The defects on the surface of these three structures were analyzed by XPS and PL spectroscopy, respectively, which also proved that tower-like structure ZnO had the highest oxygen vacancy concentration. The same is true for other MOx nanomaterials, for example, the (002) crystal plane of TiO₂ could enhance the response of the sensor^{57,192} and the (004) crystal plane of TiO₂ could realize the reduction of the response time.¹⁹³ Both the (001) crystal plane of the hexagonal WO₃¹⁸⁹ and the (110) crystal plane of SnO₂¹⁹⁰ could enhance the performance of the sensors. The regulation of the crystal plane engineering of the sensing materials provides new a research direction for the development of sensing materials in the future.



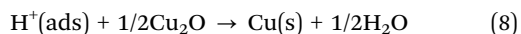
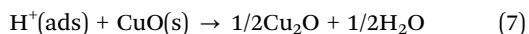
3.4 P-Type semiconductor metal oxide nanomaterials

P-Type semiconductor metal oxides are a class of materials whose conductivity decreases with the increase in the reducing gas. The application of p-type MOx in the field of gas sensing is more challenging because the main carrier of p-type MOx are the holes. Hübner *et al.*¹⁹⁴ indicated that under the same conditions, the response of p-type was equal to the square root of n-type, as depicted in eqn (6).

$$S_p = \sqrt{S_n} \quad (6)$$

The major p-type MOx nanomaterials in H₂ sensing are CuO^{111,195} and NiO,^{149,150,196} as listed in Table 6.

CuO demonstrates a promising p-type MOx nanomaterial for H₂ sensing due to the hydrogen-induced metallization phenomenon.¹⁹⁵ On the surface of CuO, the H₂ molecules not only react with the oxide anions but also react with CuO to form Cu₂O or Cu, based on the reactions (7) and (8).¹⁸⁶



High-energy radiation is also widely available in the preparation of p-type MOx. Ultraviolet irradiated CuO nanowires can detect H₂ at room temperature.¹¹¹ Pulsed laser irradiated NiO nanofilms have higher response and faster recovery time.¹⁵⁰

The choice of the sensing substrate will affect the performance of the sensor. Abubakar *et al.*¹⁴⁹ deposited a cubic NiO film on the ITO conductive glass by RF magnetron sputtering under the annealing process. The NiO film is relatively loose due to the annealing treatment. Compared with a similar compact film, the loose film means more voids. H₂ can not only react on the surface of the material but also goes deep into the interior of the material. By increasing the contact between H₂ and the material, a lower concentration can be achieved and a lower optimal operating temperature can be realized. Because its loose surface can better realize the reversible adsorption and desorption of H₂, the response/recovery time of the sensor is very fast, with a response time of 6 s and a recovery time of 0.5 s with a temperature of 150 °C, which is the fastest among the mono-semiconductor metal oxide nanomaterials.

To sum up, mono-semiconductor MOx nanomaterials as sensing materials realized the detection of H₂ with fast response, low detection limit, and high sensitivity, which can greatly improve the performance of the sensor by controlling its morphology and structure as well as regulate the oxygen vacancies on the material surface. However, the inherent limitations hamper their further improvement; one of the most critical shortcomings is the high working temperature, which seriously impedes the practical application space of the sensor. Therefore, the realization of low-power detection is spurring researchers to explore and design new functional MOx. In the following sections, mono-semiconductor MOx decoration by noble or non-noble metals and binary, ternary, or more complicated nanostructures will be discussed in detail.

4. Metal@MOx composites for H₂ gas sensor

Decorating metal nanoparticles into the MOx nanomaterials can effectively improve the performance of the gas sensors, which have received extensive attention from researchers. Decorating with ideal elements can tune the surface defects of the materials, which lead to more oxygen vacancies. Moreover, incorporating noble metal ions with high catalytic activity onto the MOx nanomaterials could form the Schottky barrier, which improves the selectivity and performance of the sensors.

4.1 Synthetic strategy for M@MOx nanocomposites

There are many methods for the synthesis of metal@MOx composites, as summarized and listed in Table 7.

Electrochemical deposition is a ubiquitously useful method for depositing a coating on the surface of the substrates.²¹³ The application of electrochemical deposition nanocomposites is more extensive and flexible for the morphology of the material and the proportion of substances in the material can be easily controlled by selecting different electrolyte solutions, different electrochemical methods, potentials, and currents during the deposition process.²¹⁴

For example, Lupan *et al.*⁶⁴ adopted a three-electrode system, in which FTO conductive glass was used as the working electrode, the deposition potential was −0.58 V, and the current density was −64 C cm^{−2}. At 91 °C, the Ag–ZnO nanoarray was obtained by

Table 6 The response capability of the p-type semiconductor metal oxide nanomaterials to H₂

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
CuO	Nano-bitter gourd	100 ppm	200	2 ppm	175% ^d	10/1016 s	59
CuO	Nanowires	100 ppm	RT	—	4.6% ^f	—	111
CuO	Nanowire	100 ppm	300	—	340% ^b	60/2 s	195
NiO	Nanoplates	100 ppm	200	—	90% ^d	180 s	60
NiO	Thin films	30 ppm	RT	30 ppm	310% ^d	6/0.5 s	149
NiO	Film	1000 ppm	175	—	32.4% ^d	134/406 s	150
NiO	Thin films	40 ppm	200	5 ppm	75% ^d	—	196
NiO	Nanosheet	150 ppm	250	10 ppm	191% ^d	150 s	60
NiO	Nanowires	50 ppm	400	—	34% ^d	35/20 s (450 °C)	197
TeO ₂	Nanowires	0.5 vol%	25/140	23 ppm	28%/57% ^d	100/250 s	61

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .



Table 7 Details about the synthesis of metal@MOx composites

Materials	Method	Precursor	Solvent	Temp. time	Conv. cond.	Ref.
Ag-ZnO	Electrodeposition	ZnCl ₂ , KCl, AgNO ₃	H ₂ O	91 °C, 200 min	250 °C, 12 h	64
Au-ZnO	Electrodeposition	ZnCl ₂ , KCl, HAuCl ₄	H ₂ O	85/90 °C	—	198
Cd-ZnO	Electrodeposition	ZnCl ₂ , KCl, CdCl ₂	H ₂ O	92 °C, 150 min	300 °C, 11 h	75
Pd-ZnO	Electrodeposition	ZnCl ₂ , KCl, PdCl ₂	H ₂ O	90 °C	250 °C, 12 h	45
Rh-SnO ₂	FSP	Tin(II) 2-ethyl hexanoate, rhodium(III) acetylacetonate	Xylene	—	—	102
Pt-SnO ₂	FSP	Tin(II) 2-ethylhexanoate, platinum(III) acetylacetonate	Xylene	O ₂ : 2.46 L min ⁻¹ , CH ₄ : 1.19 L min ⁻¹	—	199
Pt- Zn ₂ SnO ₄	FSP	Zinc(II) acetylacetonate, tin(II) 2-ethylhexanoate, platinum(II) acetylacetonate	Xylene + methanol (7 + 3)	5 mL min ⁻¹	—	200
Pd-ZnO	Spray pyrolysis deposition	Zn(Ac) ₂ ·2H ₂ O, PdCl ₂	—	—	400 °C, 1 h	201
Cr-ZnO	SPT	Zn(Ac) ₂ ·2H ₂ O Cr(NO ₃) ₃ ·6H ₂ O, HMT	H ₂ O	350 °C	450 °C, 1 h	78
Cu-CdO	SPT	Cd(Ac) ₂ , Cu(Ac) ₂	H ₂ O	300 °C	—	38
Mg-ZnO	SPT	Zn(Ac) ₂ , HMT, Mg(Ac) ₂	H ₂ O	300 °C	200 °C	202
W-ZnO	SPT	Zn(Ac) ₂ ·2H ₂ O, WCl ₆	EtOH, H ₂ O	350 °C	—	82
Pd-SnO ₂	Solvothermal	SnCl ₄ ·5H ₂ O, PdCl ₂ , NaOH	DMF	160 °C, 15 h	400 °C, 1 h	203
Co-ZnO	Hydrothermal	Zn(Ac) ₂ ·2H ₂ O, HMT, Co(NO ₃) ₂	H ₂ O	90 °C, 24 h	—	77
Cd-ZnO	Hydrothermal	Zn(NO ₃) ₂ ·6H ₂ O, HMT, Cd(NO ₃) ₂ ·4H ₂ O	H ₂ O	100 °C, 24 h	—	204
Nb-TiO ₂	Hydrothermal	TBOT, HClO ₂ , Nb(OCH ₂ CH ₃) ₅	H ₂ O	150 °C	400 °C, 20 min	205
Pd-In ₂ O ₃	Hydrothermal	InCl ₃ ·4H ₂ O, SDS, urea, PdCl ₂	H ₂ O	120 °C, 15 h	500 °C, 2 h	67
Rh-SnO ₂	Hydrothermal	RhCl ₃ ·3H ₂ O, glucose, SnCl ₄ ·5H ₂ O	H ₂ O	200 °C, 12 h	500 °C, 30 min	206
Rh/Rh _x O _y	PLD	Rhodium rod target	H ₂ O	40 min	—	207
Pd-SnO ₂	EISA	Pluronic F-127 (EO ₁₀₆ PO ₇₀ EO ₁₀₆), SnCl ₄ ·5H ₂ O, PdCl ₂	EtOH	RT, 24 h	400 °C, 3 h	208
In-ZnO	CVD	Zn powder, In powder	—	Ar: 200 sccm, O ₂ : 10 sccm	—	209
La-SnO ₂	Electrospinning	SnCl ₄ ·2H ₂ O, PVP, La(NO ₃) ₃ ·6H ₂ O	DMF, EtOH	15 kV, 20 cm	600 °C, 3 h	210
Mg-In ₂ O ₃	Electrospinning	In(NO ₃) ₃ , PVP, Mg(NO ₃) ₂ ·6H ₂ O	DMF, EtOH	15 kV, 15 cm	600 °C, 3 h	80
Pd-ZnO	Electrospinning	Zn(Ac) ₂ , PdCl ₂	PVAc, IPA	+15/-10 kV, 0.07 ml h ⁻¹ , 20 cm	600 °C, 2 h	211
Ga-ZnO	Sol-gel method	Zn(Ac) ₂ , Ga(NO ₃) ₃ , MEA	IPA	RT, 12 h	550 °C, 3 h	212
Pt@NiO	Sol-gel method	H ₂ PtCl ₆ ·6H ₂ O, stearic acid, Ni(NO ₃) ₂ ·6H ₂ O, SDBS, (NH ₄) ₂ CO ₃	EtOH	pH = 7, 60 °C, 24 h	500 °C, 2 h	47

electrochemical deposition for 200 min. The diameter of the Ag-ZnO nanoarray is about 100–400 nm and the length is about 4 μm. By adjusting the deposition potential and electrolyte solution, diverse metal-doped MOx structures such as Pd-ZnO,⁴⁵ Au-ZnO,¹⁹⁸ and Cd-ZnO⁷⁵ were also prepared for H₂ gas sensing. The electrochemical deposition technique has broad application prospects. It can not only realize the growth of semiconductor MOx and doped particles simultaneously but can also achieve surface doping and material functionalization simultaneously. Moreover, for the preparation of materials with complex structures, electrochemical deposition technology can reduce the technical steps and achieve efficient and green synthesis.

Spray pyrolysis is the process of forming aerosol from metal salt solution under the atomization of the spraying device, precipitating a solid phase due to supersaturation under high temperature, and finally forming an ultrafine powder through particle hydrolysis, sintering, or other processes.²¹⁵ With continuous development in recent decades, spray pyrolysis technology has become an important means to synthesize multi-MOx.

Spray pyrolysis has the advantages of both gas-phase and liquid-phase methods due to its special synthetic procedures.²¹⁶ Spray pyrolysis uses a liquid phase solution as the precursor; thus, the stoichiometric ratio of each component of the

composites can be precisely controlled. In the synthetic process of materials, the principle of vapor deposition is adopted, which can ensure high purity of the product.

A large number of M@MOx nanocomposites films can also be prepared by SPT.^{82,202} For example, CdO films doped with different concentrations of doped Cu were prepared on amorphous glass substrates by the spray pyrolysis technology (SPT).³⁸ In addition, SPT can also be combined with successive ionic layer deposition (SILD) to prepare noble metal-doped MOx. SILD can realize the synthesis of noble metal clusters on the surface of MOx by controlling the number of cycles of ion deposition and the thickness of deposited nanoparticles can be controlled with high precision, which is similar to ALD. Korotcenkov *et al.*⁶⁵ prepared Au-In₂O₃ thin films by SPT combined with SILD and realized the efficient detection of H₂.

It is worthwhile to mention that flame spray pyrolysis (FSP) is a promising technique to prepare MOx,²¹⁷ especially the noble metal-doped MOx nanomaterials.²¹⁸ This is because the vapor pressure of noble metals is generally higher than that of MOx; thus, MOx are first produced in flames and then noble metal nanoparticles grow heterogeneously on their surfaces *in situ*. The synthesis of Rh-SnO₂¹⁰² and Pt-SnO₂¹⁹⁹ composites with different doping amounts *via* FSP have been reported.



PLD, an emerging thin film preparation method, could realize high-quality metal–semiconductor MOx nanocomposites synthesis, whose working principle is that the plasma directed local expansion deposits a film on the substrate.²¹⁹ It has been reported that rhodium-doped nanocolloidal rhodium oxide particles can be prepared in water by nanosecond pulse excitation starting from high purity rhodium targets, which is the first application of rhodium oxide/rhodium colloids in resistive gas sensors.²⁰⁷ However, the PLD technology can cause micron–submicron particle contamination on the surface of the film during the preparation process; thus, the uniformity of the film is poor. Moreover, PLD is a high energy-consuming method and so, it is not widely used in the preparation of semiconductor MOx.

Glancing-angle deposition (GLAD) focuses on film preparation with controllable morphology.²²⁰ The schematic of glancing-angle dc magnetron sputtering deposition is shown in Fig. 10a. The morphology of Pt–WO₃ is shown in Fig. 10b. In addition to the above synthetic methods, magnetron sputtering technique can also realize the preparation of the doped MOx nanomaterials.²²¹

Hydrothermal or solvothermal technology is widely used in the preparation of metal–MOx nanocomposites. The target products can be obtained by mixing different salt solutions in the autoclave and treating them with high temperature and high pressure. By controlling the reaction conditions, MOx nanomaterials with different morphologies can be obtained. As for M@MOx nanocomposites, hydrothermal or solvothermal technology can be divided into two categories.

Wang *et al.*²²² developed a versatile two-step hydrothermal method for noble metal-doped MOx composites. Briefly, ((NH₄)₁₀–H₂(W₂O₇)₆·xH₂O) was used as the precursor and organic acid was used as the solvent to assist the synthesis of MOx nanomaterials. Then, noble metal nanoparticles were modified *in situ* on the surface of the synthesized MOx using iodide ion (I[−]) as the strong adsorbent, polyvinylpyrrolidone (PVP) as the capping agent, and *N,N*-dimethylformamide (DMF) as the solvent. Meanwhile, I[−] is also a morphology control agent during the synthetic process, which can realize the control of its

morphology. The structure of the prepared Pd–WO₃ nanoplates is shown in Fig. 11a–c. It can be seen from Fig. 11b and c that the doped Pd nanoparticles are attached to the surface of WO₃, which play an important role in improving the performance of the sensor. Liu *et al.*²²³ and Xiao *et al.*²²⁴ also successfully synthesized Pd–WO₃ nanoparticles using this method.

The hydrothermal or solvothermal method can also realize the “one-pot” preparation of M@MOx nanocomposites. This method not only shortens the reaction time but also has more advantages in terms of the economic benefits and environmental friendliness. Li *et al.*²⁰³ obtained SnO₂ nanocomposite structures loaded with Pd NPs by the “one-pot” method. Depending on the different precursors added, Cd–ZnO,²⁰⁴ Co–ZnO,⁷⁷ and other nanocomposites can also be prepared by the “one-pot” method. In the hydrothermal process, the added template plays a key role in the regulation of the material morphology. Hong *et al.*²⁰⁶ synthesized porous coral-like Rh-doped SnO₂ nanostructures using glucose as the template; the morphology is shown in Fig. 11d and e. As seen from Fig. 11f and g, the addition of Rh can alter the pore size of the materials.

In addition, the hydrothermal method combined with other methods has been widely studied and used in the preparation of M@MOx nanocomposites. The first strategy is to prepare MOx nanoparticles by the hydrothermal method and then deposit all doped noble and non-noble metal nanoparticles onto the MOx surface through vapor deposition such as CVD²²⁵ and sputtering technology.²²⁶ Similarly, the liquid phase chemical synthesis method can also be combined with it. In this synthetic strategy, MOx obtained by the hydrothermal method can be used as substrates and the dopants are directly grown on the surface of MOx *in situ* by liquid-phase chemical reduction in solution to obtain composite nanostructures.

The porosity of the materials is closely related to the gas sensing properties; thus, porous structures usually exhibit more excellent gas sensing performance than solid structures. In recent years, solution template assisted methods including soft and hard templates have produced MOx nanomaterials with ordered large surface area.

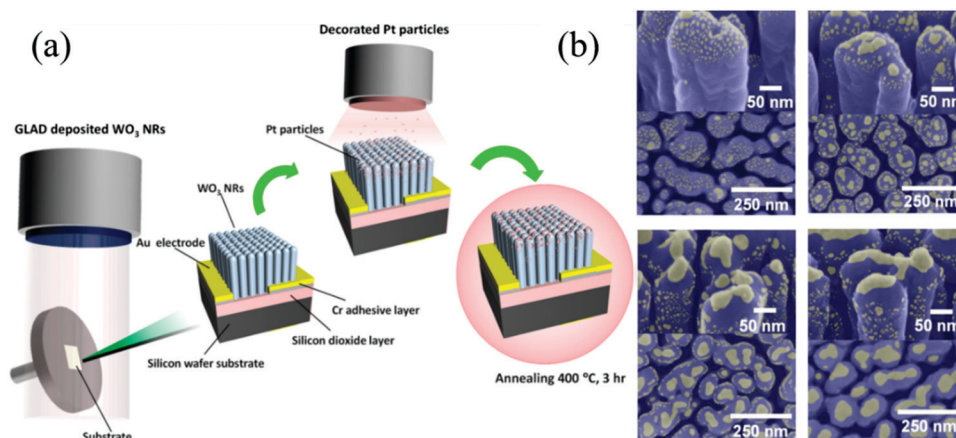


Fig. 10 (a) Schematic of the Pt-decorated WO₃ nanorod array fabricated by the glancing-angle dc magnetron sputtering deposition; (b) morphology of Pt–WO₃ doped with different ratios²²⁰ (Copyright 2014, American Chemical Society).



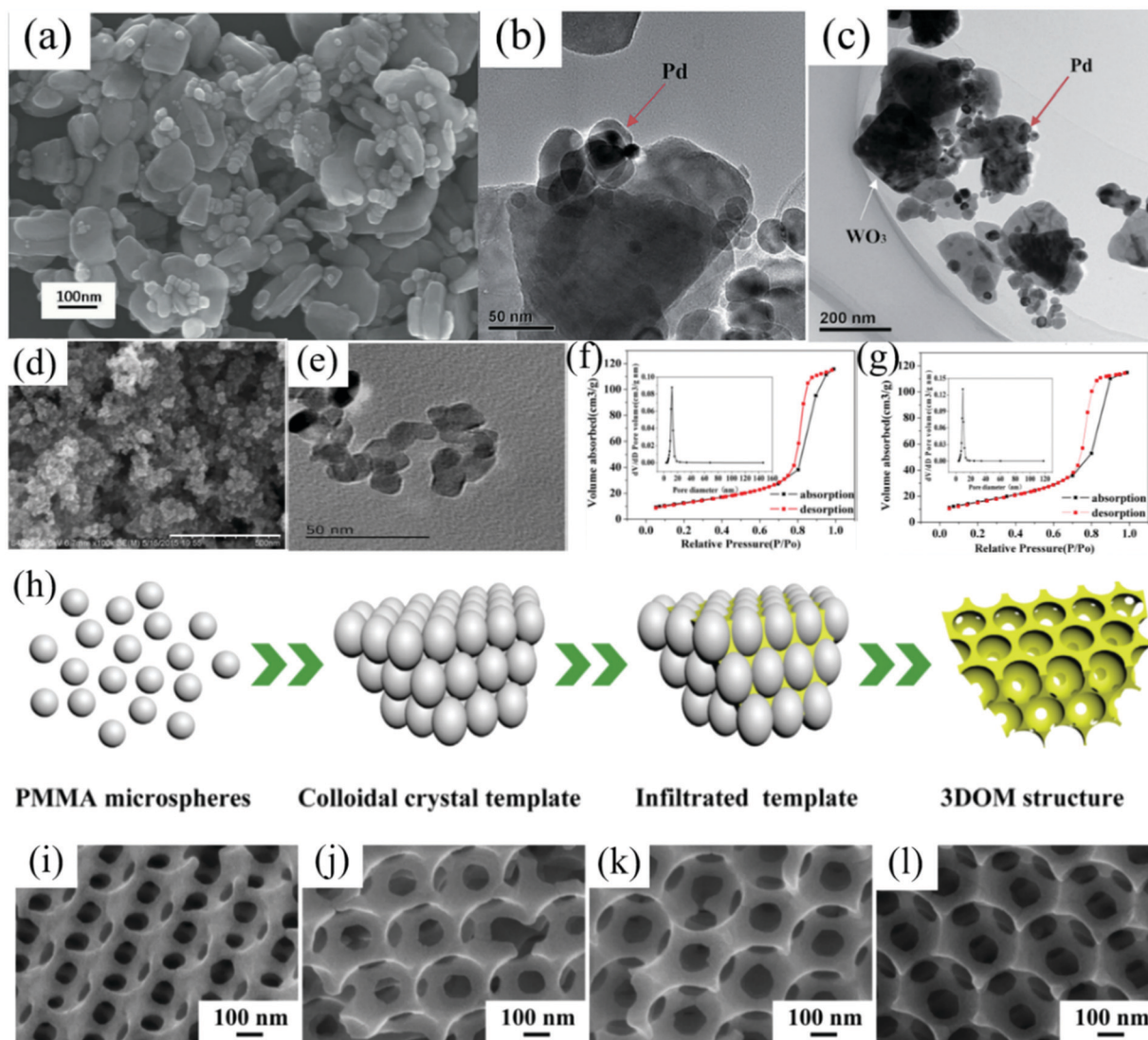


Fig. 11 (a) SEM images of Pd-WO₃; (b and c) TEM images of Pd-WO₃²²² (Copyright 2015, Royal Society of Chemistry). (d) SEM of Rh-SnO₂; (e) TEM of Rh-SnO₂; nitrogen adsorption isotherms and corresponding pore size distribution (inset) of (f) SnO₂ and (g) Rh-SnO₂²⁰⁶ (Copyright 2017, Springer Nature). (h) Schematic of the synthesis of 3DOM WO₃; SEM images of 3DOM samples: (i) WO₃; (j) Pd/WO₃ (2-1); (k) Pd/WO₃ (1-1); (l) Pd/WO₃ (1-2)⁷¹ (Copyright 2018, Elsevier).

In a typical synthetic process, Zhang *et al.*²⁰⁸ successfully fabricated multilayer mesoporous Pd-SnO₂ thin films using Pluronic F127 as the soft template by EISA. First, the Pd-doped Sn sol was prepared, then the sol was spin-coated on the Si substrate, and then the substrate was dried at room temperature with a relative humidity of 70–80% for one week. Finally, the substrate was annealed to obtain the mesoporous structure. Interestingly, Wang *et al.*⁷¹ obtained three-dimensional ordered macroporous (3DOM) WO₃ materials by liquid-phase reduction using PMMA microspheres as the hard templates. The schematic of the synthesis of 3DOM WO₃ is shown in Fig. 11h and the SEM images with different ratios of Pd and WO₃ are shown in Fig. 11i–l.

4.2 Noble metal@MOx nanomaterials for H₂ sensing

Decorating noble metals onto MOx nanomaterials is one of the most effective methods to improve the performance of H₂ gas

sensors. It can enhance the response capacity and sensitivity of the gas sensor, reduce the optimal operating temperature, and shorten the response/recovery time due to the unique nature of the noble metal nanoparticles.

Noble metal nanoparticles have stronger oxygen adsorption capacity due to the “spill-over effect”; noble metal-modified MOx can adsorb more oxygen ions and form a thick electron depletion layer on the surface of the material with higher macro-resistance. When the modified material is placed into the H₂ atmosphere, the H₂ molecules react with the oxygen anion and the electrons are transferred back to the semiconductor conduction band again, resulting in electron consumption. It causes the electron depletion layer to decrease sharply. This huge contrast between the background and the signal exceedingly reduces the signal-to-noise ratio of the sensor, and enhances the capacity of detection of trace H₂. The high catalytic activity of noble metals themselves can reduce the



activation energy required by gas adsorption, reduce the barrier required in the reaction process, and provide a possibility for the reduction of the working temperature of sensors.

Furthermore, because of the high catalytic activity, the selectivity of the sensors is considerably strengthened, especially Pd, which shows a unique property for H₂ adsorption and dissociation. Finally, the modification of noble metal ions on the surface of semiconductor MOx will accelerate the electron

transport rate on the surface of the materials and shorten the response recovery time.

The noble metal@MOx nanomaterials are widely used in H₂ gas sensing, such as Ag, Au, Pd, Pt, Rh, and their details are listed in Table 8. It is obvious that the performance of sensors in all the aspects has been greatly improved after noble metal doping.

One of the significant advantages of noble metal doping is the reduction of the working temperature, which achieves the

Table 8 The details of noble metal@MOx nanomaterials H₂ gas sensors

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
Ag-ZnO	Nanowires	100 ppm	RT	—	50 ^g	22/11 s	64
Au-In ₂ O ₃	Nanoneedles	1000 ppm	450	—	8.5 ^a	—	65
Au-In ₂ O ₃	Core-shell NPs	100 ppm	300	2 ppm	34.38 ^a	31 s/10 min	227
Au-SnO ₂	Nanoparticles	100 ppm	250	1 ppb	25 ^a	1/3 s	49
Au-ZnO	Core-Shell structure	100 ppm	300	0.5 ppm	103.9 ^a	—	50
Au-ZnO	Films	1000 ppm	250	50 ppm	172 ^e	4/68 s	66
Au-ZnO	Thin films	1000 ppm	150	—	~ 75% ^d	—	151
Au-ZnO	Nanowires	20 ppm	RT	20 ppm	2.25 ^c	—	198
Au-ZnO	Thin films	1200 ppm	400	75 ppm	73% ^e	—	228
Pd-In ₂ O ₃	Flower-like	100 ppm	210	10 ppm	3.6 ^a	4/7 s	67
Pd-In ₂ O ₃	Nanoparticles	10 000 ppm	250	150 ppm	3526 ^a	2/180 s	229
Pd-MnO ₂	Nanowalls	100 ppm	100	10 ppm	11.4 ± 0.7 ^a	4 s	37
Pd-SnO ₂	Microspheres	1000 ppm	200	10 ppm	129.08 ^a	4 s	203
Pd-SnO ₂	Mesoporous film	1000 ppm	275	50 ppm	237.85 ^a	44 s	208
Pd-SnO ₂	Thin film	250 ppm	300	25 ppm	28 ^a	3/50 s	230
Pd-SnO ₂	Nanofiber rods	100 ppm	160	0.25 ppm	28 ^a	4 s	231
Pd-Sn(Sb)O ₂	Nanoparticles	0.1 vol%	250	—	85.5% ^d	120 s	232
Pd-TiO ₂	Pd: nanocubes	0.6 vol%	150	—	40.6% ^d	24/1 s	233
	TiO ₂ : nanowire						
Pd-TiO ₂	Nanotubes	8000 ppm	RT	5000 ppm	92.05 ^d	3.8/43.3 s	68
Pd-TiO ₂	Nanorods	1000 ppm	200	250 ppm	31 ^f	—	225
Pd-V ₂ O ₅	Thin films	100 ppm	100	2 ppm	5.7 ± 0.3 ^a	—	69
Pd-W ₁₈ O ₄₉	Urchin-like	50 ppm	100	50 ppm	32 ^a	60 s	70
Pd-WO ₃	Ordered macroporous	50 ppm	130	10 ppm	382 ^a	10/50 s	71
Pd-WO ₃	Thin film	2 vol%	80	100 ppm	1.30 × 10 ^{4e}	< 1/30 s	234
Pd-WO ₃	Nanoplates	0.1 vol%	80	—	169.3 ^b	42.8/48.5 s	222
Pd-WO ₃	Nanocomposites	1000 ppm	RT	—	34 ^a	24 s	223
Pd-WO ₃	Nanotubes	500 ppm	450	—	17.6 ^a	25 s	235
Pd-WO ₃	Nanoneedles	500 ppm	150	—	1670 ^a	—	236
Pd-ZnO	“Nanosponge” film	2 vol%	80	0.1 vol%	580 ^a	0.3/22 s	36
Pd-ZnO	Nanowires	100 ppm	RT	—	13100 ^c	6.4/7.4 s	45
Pd-ZnO	Nanowire	400 ppm	RT	100 ppm	121 ^a	—	201
Pd-ZnO	Nanofibers	100 ppb	350	0.1 ppm	74.7 ^a	—	211
Pd-ZnO	Nanorods	500 ppm	350	—	3.6 ^a	—	226
Pd-ZnO	Nanorod	1 vol%	80	100 ppm	7950 ^f	227/95 s	237
Pd-ZnO	Nanowires	100 ppm	350	0.1 ppm	87.17 ^a	—	238
Pd-ZnO	Nanorods	250 ppm	135	50 ppm	22.5 ^a	26/5 s	239
Pd-ZnO	Nanorods	1000 ppm	RT	0.2 ppm	91.2 ^a	18.8 s	240
Pd-ZnO	Pd: nanocubes	10 000 ppm	100	1 ppm	0.74 ^d	—	241
	ZnO: nanorods						
Pt-In ₂ O ₃	Nanocubes	15 000 ppm	RT	—	20 ^a	33/66 s	62
Pt-Nb ₂ O ₅	Porous ceramics	10 000 ppm	RT	—	165 ^a	7/39 s	63
Pt-NiO	Core-shell structure	5000 ppm	RT	1000 ppm	4.25 ^a	91/8 s	47
Pt-SnO ₂	Nanosheets	100 ppm	350	0.08 ppm	56.5 ^a	29 s	72
Pt-SnO ₂	Thin film	500 ppm	110	2 ppm	168 ^e	< 6/57 s	242
Pt-SnO ₂	Thin film	250 ppm	200	25 ppm	51.6 ^a	—	243
Pt-TiO ₂	Nanofibers	700 ppm	—	100 ppm	400% ^f	—	73
Pt-TiO ₂	Nanocomposites	1000 ppm	RT	30 ppm	6000 ^a	10/20 s	244
Pt-TiO ₂	Thin film	10 000 ppm	RT	300 ppm	1.58 × 10 ^{7a}	150/280 s	245
Pt-ZnO	Thin film	75 ppm	300	75 ppm	60% ^d	—	7
Rh-Rh _x O _y	Film	50 ppm	200	75 ppm	23 ^e	2/5 min	207
Rh-SnO ₂	Nanoparticles	3 vol%	300	—	22170 ^a	6 s	102
Rh-SnO ₂	Coral-like	100 ppm	260	10 ppm	312 ^a	—	206

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .



detection of H_2 at low temperature or even room temperature. Because of the catalytic ability of noble metals, the activation energy during the reaction between materials and H_2 is greatly reduced, which could further reduce the activation barrier in the reaction between the materials and H_2 , thereby reducing the working temperature of the gas sensor. Vijayalakshmi *et al.*²⁰¹ prepared Pd-ZnO nanowires to detect H_2 at room temperature and the sensor responded to H_2 at 400 ppm up to 121 (defined as R_a/R_g). Ag-ZnO nanostructures prepared by the electrochemical deposition by Lupan *et al.* could also achieve efficient detection at room temperature.⁶⁴

The response of the sensor highly depends on the interaction between the target gas and the surface of the material. Doping noble metal particles is quite a powerful method to enhance the capacity of the surface of the material to adsorb oxygen and it is extensively carried out to improve the response of the sensor as well as to reduce the detection limit of the sensor. Hong *et al.*²⁰⁶ synthesized Rh-doped coral SnO_2 nanostructures by the hydrothermal method. The sensor exhibits a high response of 312 (defined as R_a/R_g) to 100 ppm H_2 at the optimum operating temperature, which is 26 times higher than that of the undoped coral SnO_2 and the detection limit of Rh- SnO_2 for H_2 is only 10 ppm. The composite can adsorb more reactive oxygen molecules and improve the sensing response owing to the high electron density of Rh.

Moreover, the sensor based on Pd- V_2O_5 can realize the sensitive and rapid detection of H_2 .⁶⁹ Pd doping can generate active centers on its surface, which has an advantage over the adsorption of target gases and the improvement of electron

transport. Moreover, Pd has a higher work function and high catalytic activity, which can bring electronic and chemical sensitization on the surface of MO_x , and improve the sensor response capacity (as shown in Fig. 12a and b). In addition, the prepared Pd- V_2O_5 films have excellent stability and selectivity. Pd- V_2O_5 can maintain a high response capacity towards H_2 for 3 months. Compared with other interfering gases, Pd- V_2O_5 has the highest response value to H_2 under the optimal working conditions, which is highly satisfactory.

After noble metal modification, the response/recovery time of the sensor is also greatly improved. Chen *et al.*⁶⁷ prepared flower-shaped spherical Pd- In_2O_3 nanostructures by the hydrothermal method. The response speed is shortened and the sensitivity is enhanced. It is because after Pd incorporation, a large number of high active reaction sites and oxygen functional groups on the surface of MO_x are produced, and the activation energy is minimized due to the high catalytic activity of the Pd nanoparticles, which accelerates the occurrence of the reaction.

Moreover, the effect of the morphology on the sensor performance is also applicable to noble metal-modified MO_x . Tang *et al.*²³⁹ explored the effect of Pd NPs with different morphologies (Pd nanoparticles are cubical, spherical, and octahedral) on the gas sensing performance of Pd-ZnO nanorods. From the test results, the cubic Pd-doped ZnO nanorods have the highest sensing response capacity (shown in Fig. 12b); thus, the morphology of the doped noble metal nanoparticles will also have an impact on the sensing characteristics, which has a certain guiding significance for the preparation of sensing materials.

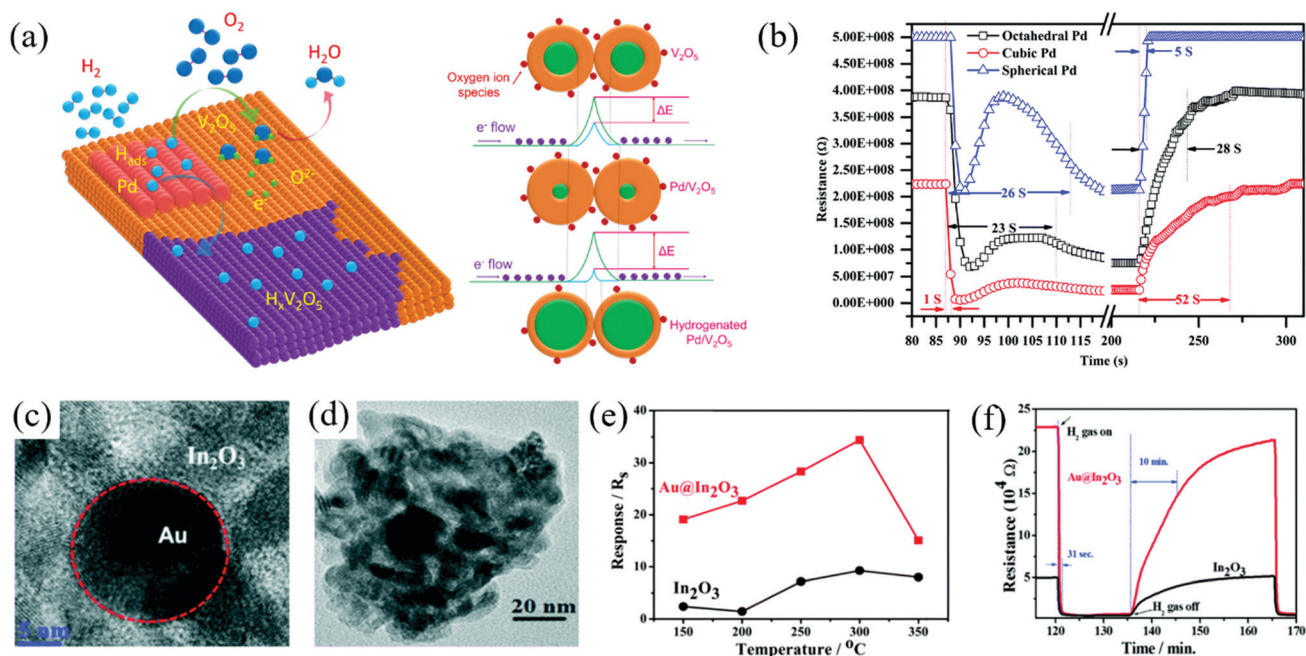


Fig. 12 (a) Schematic illustration explaining the H_2 sensing mechanism of the Pd/ V_2O_5 thin film sensor⁶⁹ (Copyright 2016, Elsevier). (b) Response curve and response/recovery time of Pd-ZnO composites with different morphologies to the H_2 test chart²³⁹ (Copyright 2018, Royal Society of Chemistry). (c and d) HR-TEM images of the Au/ In_2O_3 core-shell structure; (e) dependence of the sensor response of Au@ In_2O_3 core-shell and In_2O_3 NPs at different temperatures to 100 ppm H_2 gas; (f) dynamic response-recovery curves at an operating temperature of 300 °C and 100 ppm H_2 gas²²⁷ (Copyright 2016, Royal Society of Chemistry).



In order to enhance the performance of the sensors, the manner of noble metal doping has gradually changed. In most cases, noble metal particles are dispersed on the surface of MOx but sometimes, noble metals may occupy the active sites on the surface of MOx. Moreover, owing to the generally small size of noble metal nanoparticles, the phenomenon of agglomeration on the oxide surface may occur, thereby attenuating the sensor performance.

Designing core-shell structures can relieve this phenomenon due to the unique structural advantages. The existence of the core-shell structure can protect the noble metal core to the greatest extent; furthermore, the characteristics of the three-dimensional structure of the core-shell structure are more conducive for the transmission of electrons between the interfaces and to improve the response of the sensors.²⁴⁶ Chava *et al.*²²⁷ synthesized the Au/In₂O₃ core-shell structure by the hydrothermal method, followed by the chemical bath method. As shown in Fig. 12c and d, the performance of the Au/In₂O₃ core-shell structure is excellent at the optimum working temperature, and its response value and selectivity are greatly improved compared with In₂O₃ (Fig. 12e). In comparison with the other structures, the core-shell structure has the advantage of working temperature, response, and so on. However, due to the existence of its core-shell structure, complete H₂ desorption on the material surface cannot be achieved. Therefore, in the test process, the recovery time of the sensor is sluggish (Fig. 12f); sometimes, the phenomenon of baseline drift also occurs.

High-energy radiation still plays an unparalleled role in improving the performance of sensors based on noble metal-decorated MOx nanomaterials. The principle of high energy radiation for improving the performance of sensitive materials has been discussed in Section 3.2.1; thus, it will not be repeated in this section.

For instance, under high-energy radiation, the stability of the gas sensors is improved and the response/recovery time is shortened. Zhao *et al.*³⁶ activated the prepared Pd-ZnO sponge-type films under ultraviolet light at 365 nm; it can be seen from Fig. 13c and d that the stability of the gas sensors has been developed and the response time is only 0.3 s. Kim *et al.*²¹¹ prepared Pd-ZnO nanofibers by the electrospinning technology and activated the material under high energy radiation. As can be seen from Fig. 13a and b, with the increase in the high energy radiation, the performance of the sensor showed an upward trend and its detection limit could reach 0.1 ppm.

In conclusion, noble metals could lower the working temperature, improve the sensitivity and the selectivity, as well as shorten the response/recovery time because of the synergistic or additive effect of chemical and electronic sensitization. The high catalytic activity and fast carrier transport could enhance the maximum potential of the MOx nanomaterials in H₂ sensing. However, the high cost impedes the commercialization application. Therefore, to reduce the cost of sensing materials, searching for low-cost MOx nanocomposites will provide technical support for the industrialization and commercialization of sensors.

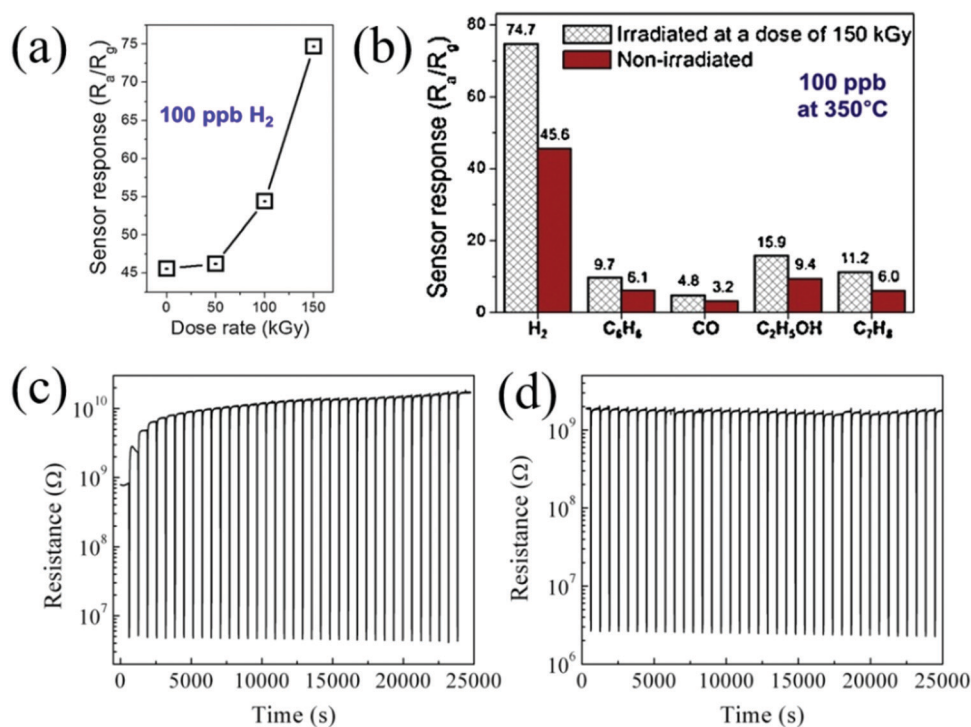


Fig. 13 (a) Sensor response versus dose rate; (b) comparison of the selectivity of non-irradiated and 150 kGy-irradiated 0.6 wt% Pd-loaded ZnO NF sensors²¹¹ (Copyright 2019, Elsevier). (c) Stability of gas sensors without UV irradiation; (d) stability of gas sensors under UV irradiation³⁶ (Copyright 2017, Elsevier).



4.3 Non-noble metal@MOx nanomaterials for H₂ sensing

Although noble metal-decorated MOx nanomaterials have excellent sensing response, their high costs limit the large-scale application of sensors. Inspired by noble metal doping, researchers have paid attention to the decoration of other metals. The performance of H₂ gas sensing is boosted by doping metal ions, such as Cd, Co, Mg, Nb, La, Cr, W, Al, Eu, In, Ni, and Cu, as listed in Table 9.

Doping non-noble metal(s) into MOx is promising. Firstly, doping could lead to a significant change in the grain size of MOx. It changes the specific surface area of the material, which can increase the effective sensing area of the nanocomposites. A large specific surface area means more gas adsorption sites, which can enhance the sensing response capacity. Secondly, elemental doping is the incorporation of elements into the lattice of MOx in the form of ion substitution. The decorated particles will replace the position of the original metal particles and cause lattice distortion. These distortions will lead to an increase in the surface defects of the materials. The presence of surface defects will introduce a large number of oxygen vacancies, which leads to an increase of the depletion layer on the surface of the composite material and a higher barrier, making the sensor more sensitive to the targeted gas and reducing the detection limit of the sensor. Finally, the presence of other metal ions can adjust the energy band structure. By modifying the MOx by changing the surface potential and the carrier concentration in the composite material, rapid charge transfer could be achieved and the detection range of the sensor can be increased.

To sum up, heteroatom doping can improve the performance of the sensor due to the ability of interfacial and surficial electron tuning. The working temperature and responsiveness of the gas sensor can also be effectively ameliorated.

5. Binary metal oxide nanostructures for H₂ gas sensors

Binary metal oxide nanostructures represent one of the most promising gas sensing materials owing to the existence of heterostructures. In H₂ gas sensing, MOx can be combined with MOx to form MOx nanoscale heterostructures or combined with carbon-based materials to obtain the composites to enhance the performance of the sensors. In this section, after describing the synthesis of binary MOx nanomaterials, MOx nanoscale heterostructures and MOx-carbon nanocomposites for H₂ sensing are presented.

5.1 MOx nanomaterial heterostructure

5.1.1 Synthetic strategy for MOx nanomaterial heterostructure. The overall structure of nanocomposites is a considerable factor that determines their properties. The choice of synthetic methods and the design of synthetic routes will have a vital impact on the morphology and stoichiometric ratio of the composites. Therefore, choosing a suitable synthetic strategy is of great interest to improve the sensing performance of MOx heterostructures.

To achieve precise control of the composition and morphology of MOx heterostructures, there is a variety of synthetic strategies applied to the construction of heterostructures. With the continuous maturation of material preparation technology, MOx heterostructures with various morphologies and structures have been prepared for the H₂ gas sensor. The synthetic methods of MOx heterostructures mainly include electrospinning technology, hydrothermal method, and thermal oxidation method, as listed in Table 10.

With the development of electrospinning technology, equipment, and application, various types of electrospinning fiber

Table 9 The response of non-noble metal@MOx nanomaterials to H₂

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
Al-ZnO	Thin films	1000 ppm	300	100 ppm	66% ^d	—	74
Cd-ZnO	Nanowires	100 ppm	RT	—	3.7 ^d	—	75
Cd-ZnO	Nanorods	1000 ppm	80	50 ppm	6 ^a	43 s	204
Co-SnO ₂	Nearly spherical	2000 ppm	250	—	100 ^a	3/15 s	144
Co-SnO ₂	Nearly spherical	2000 ppm	225	—	59.4 ^a	7 s	76
Co-ZnO	Nanorods	3000 ppm	300	—	99.2% ^f	74/40 s	77
Cr-ZnO	Nano-whiskers array	400 ppm	RT	75 ppm	133 ^e	—	78
Cu-CdO	Cubic structure	100 ppm	300	100 ppm	48.00% ^e	—	38
Eu-SnO ₂	Nanoparticles	300 ppm	350	100 ppm	~20 ^a	7 s	79
Eu-SnO ₂	Columnar films	100 ppm	250	—	115 ^c	—	247
Er-SnO ₂	Nanoparticles	100 ppm	360	—	28 ^a	11/42 s	248
In-ZnO	Thin film	5 ppm	300	1 ppm	15% ^d	—	249
La-SnO ₂	Nanofiber	100 ppm	300	5 ppm	9.9 ^a	1/1 s	210
Mg-In ₂ O ₃	Nanotubes	100 ppm	150	—	1.55 ^a	—	80
Mg-ZnO	Like-spherical	400 ppm	RT	150 ppm	~2.75	—	202
Mg-ZnO	Nanorods	200 ppm	RT	—	30 ^a	—	250
Mg-ZnO	Film	200 ppm	RT	100 ppm	35–40 ^a	75/54 s	251
Nb-TiO ₂	Nanotubes	1000 ppm	RT	50 ppm	30.9 ^d	100 s	81
Nb-TiO ₂	Nanorods	1 ppm	RT	1 ppm	22.5% ^d	288/324 s	205
Ni-TiO ₂	Nanospheres	300 ppm	RT	100 ppm	~220 ^a	40 s	252
Ni-ZnO	Thin film	5 ppm	150	—	17.77% ^d	—	253
W-W ₁₈ O ₄₉	Core-shell nanoflowers	—	—	50 ppm	—	—	254

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .



Table 10 Details of MOx nanomaterial heterostructure synthesis

Materials	Method	Precursor	Solvent	Temp., time	Conv. cond.	Ref.
SnO ₂ -ZnO	Electrospinning	SnCl ₂ ·2H ₂ O, Zn(OAc) ₂ , PVP	DMF, EtOH	15 kV, 20 cm, 0.05 mL h ⁻¹	700 °C, 0.5 h	255
SnO ₂ -ZnO	Electrospinning	SnCl ₂ ·2H ₂ O, Zn(OAc) ₂ , PVP	DMF, EtOH	15 kV, 20 cm, 0.05 mL h ⁻¹	700 °C, 0.5–24 h	256
ZnO-NiO	Electrospinning	ZnCl ₂ ·2H ₂ O, Ni(Ac) ₂ ·4H ₂ O	PVA	+15/-10 kV, 20 cm, 0.01 mL h ⁻¹	600 °C, 2 h	257
SnO ₂ -ZnO	Electrospinning	ZnCl ₂ ·2H ₂ O, SnCl ₂ ·2H ₂ O	PVA	+15/-10 kV, 20 cm, 0.01 mL h ⁻¹	600 °C, 2 h	187
Co ₃ O ₄ -ZnO	Electrospinning	ZnCl ₂ ·2H ₂ O, Co(Ac) ₂ ·4H ₂ O, PVA	H ₂ O	+15/-10 kV, 20 cm, 0.01 mL h ⁻¹	600 °C, 2 h	84
NiO-SnO ₂	Electrospinning	NiCl ₂ ·6H ₂ O, PVP, SnCl ₂ ·2H ₂ O	EtOH, DMF	12 kV, 15 cm, 1 mL h ⁻¹	500 °C, 3 h	258
CeO ₂ -SnO ₂	Hydrothermal	Ammonium cerium(IV) nitrate, Sn powder, NH ₃ ·H ₂ O	H ₂ O	200 °C, 10 h	500 °C, 2 h	259
SnO-SnO ₂	Hydrothermal	Sn(Ac) ₂ , PdCl ₂ , NaOH, PEG-400, EtOH	H ₂ O	140 °C, 24 h	—	39
CeO ₂ -In ₂ O ₃	Hydrothermal	In(NO ₃) ₃ ·4.5H ₂ O, Ce(NO ₃) ₃ ·6H ₂ O, urea	H ₂ O	160 °C, 12 h	500 °C, 2 h	85
ZnO-Nb ₂ O ₅	Sol-gel method	Zn(Ac) ₂ , PEG, NH ₃ ·H ₂ O	H ₂ O	—	—	260
Zn ₂ SnO ₄ -ZnO	Thermal evaporation	SnO ₂ powders, ZnO powders, C powders	—	97 vol% Ar + 3 vol% O ₂ 100 mL min ⁻¹	1000 °C, 1 h	261

materials are constantly emerging; thus, the electrospinning technology has become one of the most active methods to prepare MOx heterostructures.²⁶² Katoch *et al.*^{255,256} used electrospinning to synthesize a series of SnO₂-ZnO nanocomposite fibers with different ratios ($x\text{ZnO}-(1-x)\text{SnO}_2$ ($x = 0.01-0.50$),²⁵⁵ $0.9\text{SnO}_2-0.1\text{ZnO}$ ²⁵⁶) and with morphology control of the nanofibers. The annealing process is validated as an important process in the synthesis of the nanofibers by electrospinning as it can not only remove the residual organic solvents in the nanofibers to purify the nanofibers but can also increase the porosity of the nanofibers with interesting morphology to improve the sensing performance of the materials.²⁵⁸ A large number of MOx nanofiber heterostructures have been synthesized by the electrospinning technology, such as ZnO-Co₃O₄,⁸⁴ NiO-ZnO,²⁵⁷ and SnO₂-ZnO.¹⁸⁷

Besides the electrospinning method, thermal evaporation is a fast-developing technique for the preparation of MOx heterostructures by multi-step synthesis. Combined with the hydrothermal method, Co₃O₄-WO₃,²⁶³ and ZnO-WO₃²⁶⁴ heterostructures were obtained. Combined with the solvothermal deposition technology, the Nb₂O₅-TeO₂ heterostructure was obtained.⁸⁷

The core-shell structure could reap huge fruits because of the more effective surface areas for gas adsorption. The LD technology can deposit uniform coatings on micro/nanostructured substrates with high specific surface area, especially nanowires and nanorod core-shell structures. Park *et al.*²⁶⁵ combined thermal evaporation with the ALD technology to obtain ZnO-Nb₂O₅ nanorods with a core-shell structure. Further, SnO₂-NiO core-shell nanowires were also prepared by the ALD technology.²⁶⁶ Apart from the ALD technology, it is noted that Xun *et al.*²⁶⁶ obtained highly-ordered SnO₂-TiO₂ nanotubes by the process of anodization and impregnation.

In addition, hydrothermal or solvothermal methods are beneficial for preparing the MOx heterostructure. The preparation is similar to the metal@MOx composites and will not be elaborated in depth here.

5.1.2 MOx nanoscale heterostructures for H₂ sensing. At the micro-scale, the physical interface between the two different materials is called the heterostructure, which combines two different components. Because different MOx have different bandgaps and work functions, rapid electron transfer will occur at the heterojunction, leading to high-energy level lateral shape.

A depletion layer is formed at the high energy level side and a cumulative layer is formed at the low energy level side, respectively. After contacting with H₂, the resistance of the composite changes obviously and the resistance modulation ability of the composite material is improved, thereby improving the performance of the sensor. Heterojunctions are divided into p-n junctions, n-n junctions, p-p junctions, and other different structures according to the n-type and p-type of semiconductors. The gas sensing properties of different heterostructures are listed in Table 11.

(i) *N-N heterostructure.* As early as 2013, Mondal *et al.*⁸³ synthesized the ZnO-SnO₂ n-n heterostructure by the solvothermal method and annealing to obtain ZnO nanorods with a diameter of ~150 nm loaded with 50–90 nm SnO₂ nanospheres. ZnO and SnO₂ are both n-type semiconductors; the n-n heterojunction is formed between ZnO and SnO₂. Due to the existence of the heterojunction, the sensor's response capacity to H₂ and the response recovery time are greatly improved. In the heterojunction, electrons can transfer from SnO₂ ($\Phi = 5.2$ eV) to ZnO ($\Phi = 4.9$ eV). In the prepared ZnO-SnO₂ composite structure, two depletion layers are formed, one on the surface of a single grain and the other on the heterojunction surface of ZnO-SnO₂. The presence of two depletion layers promotes the adsorption of a higher concentration of oxygen on the sensor surface to a greater extent, which provides more reaction sites for H₂. Therefore, the sensor exhibits a higher resistance in the air and the change in the resistance becomes more obvious when exposed to H₂, leading to a greatly improved sensor response.

Katoch *et al.*^{255,256} synthesized ZnO-SnO₂ nanofibers by the electrospinning technology and proposed a bifunctional sensing mechanism of the composite nanofibers. The basic principle of this mechanism is to improve the performance of H₂ sensor based on the combination of the homogeneous interface of SnO₂-SnO₂ and the heterogeneous interface of ZnO-SnO₂ (Fig. 14a and b). In the research process, Katoch *et al.* optimized the ratio of ZnO and SnO₂. By preparing different ratios of the ZnO-SnO₂ composite and testing under the same conditions, it was found that when the ratio of Sn : Zn was 9 : 1, the sensor had the greatest performance, which can detect H₂ at even 0.1 ppm concentration (Fig. 14c–e). This is because when the ratio is 9 : 1,



Table 11 The response of heterostructures to H₂

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
CeO ₂ -In ₂ O ₃	Hollow spheres	50 ppm	160	0.01 ppm	20.65 ^a	1/9 s	85
CeO ₂ -SnO ₂	Nanosheets	0.5/60 ppm	300	0.5 ppm	82/1323 ^a	—	259
Co ₃ O ₄ -SnO ₂	Nanoparticles	50 ppm	300	5 ppm	20% ^d	—	48
Co ₃ O ₄ -ZnO	Nanofibers	10 ppm	300	1 ppm	113.65 ^d	70 s	84
Co ₃ O ₄ -WO ₃	Nanowires	2000 ppm	200	—	610% ^a	—	263
Cr ₂ O ₃ -Nb ₂ O ₅	Nanocubes	2 ppm	RT	—	5.24 ^a	40 s	267
CuO-TiO ₂	Nanotubes	1000 ppm	200	—	2 ^f	7.4/6.8 s	46
CuO-Nb ₂ O ₅	Nanorods	0.5 vol%	300	0.5 vol%	217.05% ^b	161-199/163-171 s	86
In ₂ O ₃ -ZnO	Thin films	50 ppm	300: n type 200: p type	—	n: 41% ^d p: 2.5% ^d	n: 53/79 s p: 119/123 s ~50/~90 s	268
NiO-ZnO	Nanofibers	10 ppm	200	1 ppm	~60 ^d	—	257
Nb ₂ O ₅ -NiO	Nanoparticles	0.05 vol%	RT	—	1.68 ^a	—	29
Nb ₂ O ₅ -TeO ₂	Nanobelts	10 000 ppm	175	100 ppm	10.22 ^a	—	87
Nb ₂ O ₅ -ZnO	Nanorods	100 ppm	300	100 ppm	156% ^a	21 s	265
SnO-SnO ₂	Nanorods	100 ppm	50	100 ppm	1.2 ^a	5/45 s	39
SnO ₂ -NiO	Nanofibers	100 ppm	195	1 ppm	37.15 ^a	12/5 s (25 ppm H ₂)	258
SnO ₂ -NiO	Nanowires	500 ppm	500	—	114 ^d	—	269
SnO ₂ -TiO ₂	Nanotubes	1000 ppm	250	20 ppm	1410 ^a	—	266
SnO ₂ -WO ₃	Film	1000 ppm	225	50 ppm	29.31 ^a	8.4 s	270
SnO ₂ -ZnO	Hexagonal rod	10 000 ppm	150	3000 ppm	90% ^d	60 s	83
SnO ₂ -ZnO	Nanofibers	10 ppm	300	—	168.6 ^a	—	255
SnO ₂ -ZnO	Nanofibers	10 ppm	350	0.1 ppm	98.3 ^a	138/221 s	256
SnO ₂ -ZnO	Nanofibers	5 ppm	300	0.05 ppm	91 ^a	~25/~30 s	257
ZnO-WO ₃	Nanowires	5000 ppm	200	100 ppm	12.6 ^a	—	264
ZnO@ZIF-8	Nanorod film	50 ppm	250	5 ppm	3.28 ^a	—	271

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .

the grain size of the composite heterojunction is the smallest (Fig. 14f).

Moreover, if the content of ZnO is very high, contact will occur between the ZnO nanoparticles and metal-metal contact

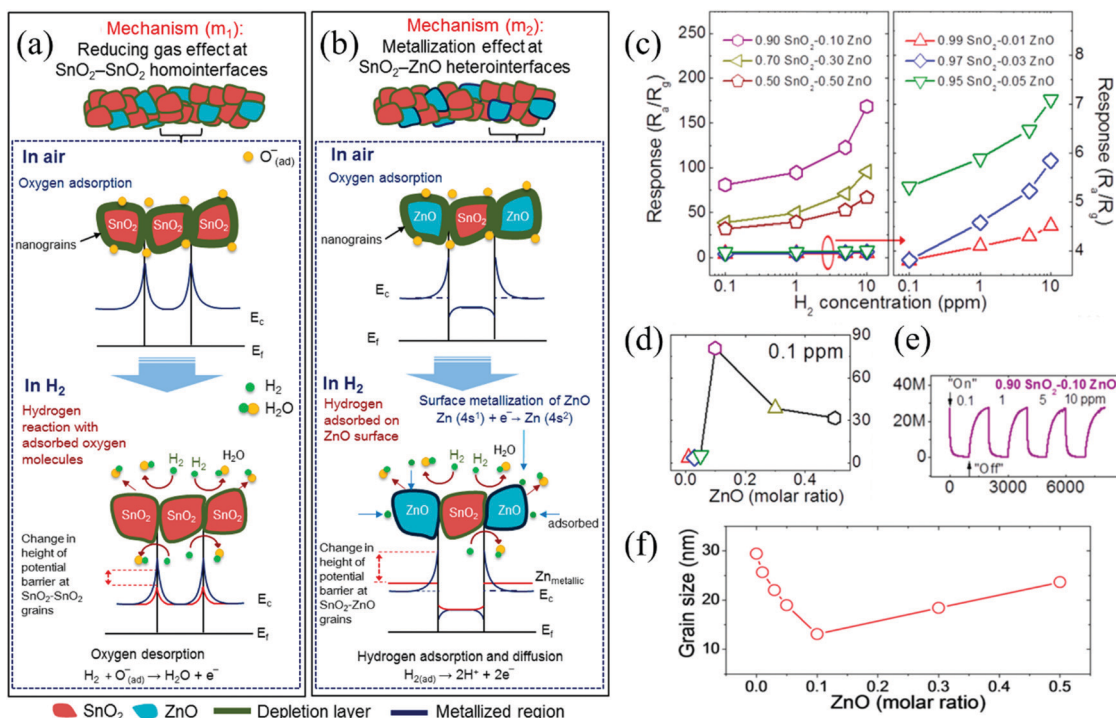


Fig. 14 (a) Mechanism 1: reducing gas effect at the SnO₂-SnO₂ homo interfaces; (b) mechanism 2: metallization effect at the SnO₂-ZnO heterointerfaces; (c) summary of the sensor responses of the SnO₂-ZnO composite nanofibers; (d) sensor responses of the SnO₂-ZnO composite nanofibers to 0.1 ppm of H₂ with different ZnO contents; (e) the response of SnO₂-ZnO nanocomposites for 0.1-10 ppm of H₂; (f) the grain size of the different nanofibers analyzed in this study²⁵⁵ (Copyright 2015, American Chemical Society).



will be formed between two adjacent ZnO nanoparticles, which reduces the modulation ability of the resistance of the nano-composites. On the other hand, if Zn is too low, the number of ZnO-SnO₂ heterojunctions will be reduced, the effect of metal-ization will also be reduced, and the resistance modulation ability will also be reduced. The proposed bifunctional sensing mechanism provides theoretical support for the preparation and application of ZnO-based heterostructures in the future.

An interesting phenomenon was found by Pati *et al.*,²⁶⁸ in which In₂O₃-ZnO thin films prepared by the sol-gel method could undergo transition from n-type to p-type during the detection of the gas at a certain temperature. Below 200 °C, the sensor produces a p-type response to H₂, while above 200 °C, the sensor produces an n-type response to H₂. The response diagram of the sensor to the same concentration of H₂ at 200 °C and 300 °C is shown in Fig. 15a and b.

This phenomenon occurs due to the reversal of carriers. Since the surface conduction of semiconductor oxides is caused

by the joint action of electrons and holes, most of the changes in the carrier density will lead to the reversal of the type of movable carriers on the surface. When the electron concentration is less than the hole concentration, the transition from “n” to “p” occurs.

Researchers also regulated the micro-morphology of the n-n heterostructure to obtain a better sensing response. Hu *et al.*⁸⁵ synthesized 0.1–1.5 μm CeO₂-In₂O₃ *via* the hydrothermal method. The structure of In₂O₃ hollow microspheres and its SEM images are shown in Fig. 15c. Compared with other nanostructures, the three-dimensional hierarchical system can significantly enhance the sensing characteristics of the gas, which is conducive for gas adsorption and desorption. The high specific surface area of the hollow structure and the synergistic effect of the heterogeneous structure make the sensor exhibit excellent H₂ sensing ability, along with good repeatability and long-term stability (Fig. 15e). The detection limit of the sensor can reach 10 ppb (Fig. 15f) and the response recovery time is only 1/9 s (Fig. 15d). The sensor is tested

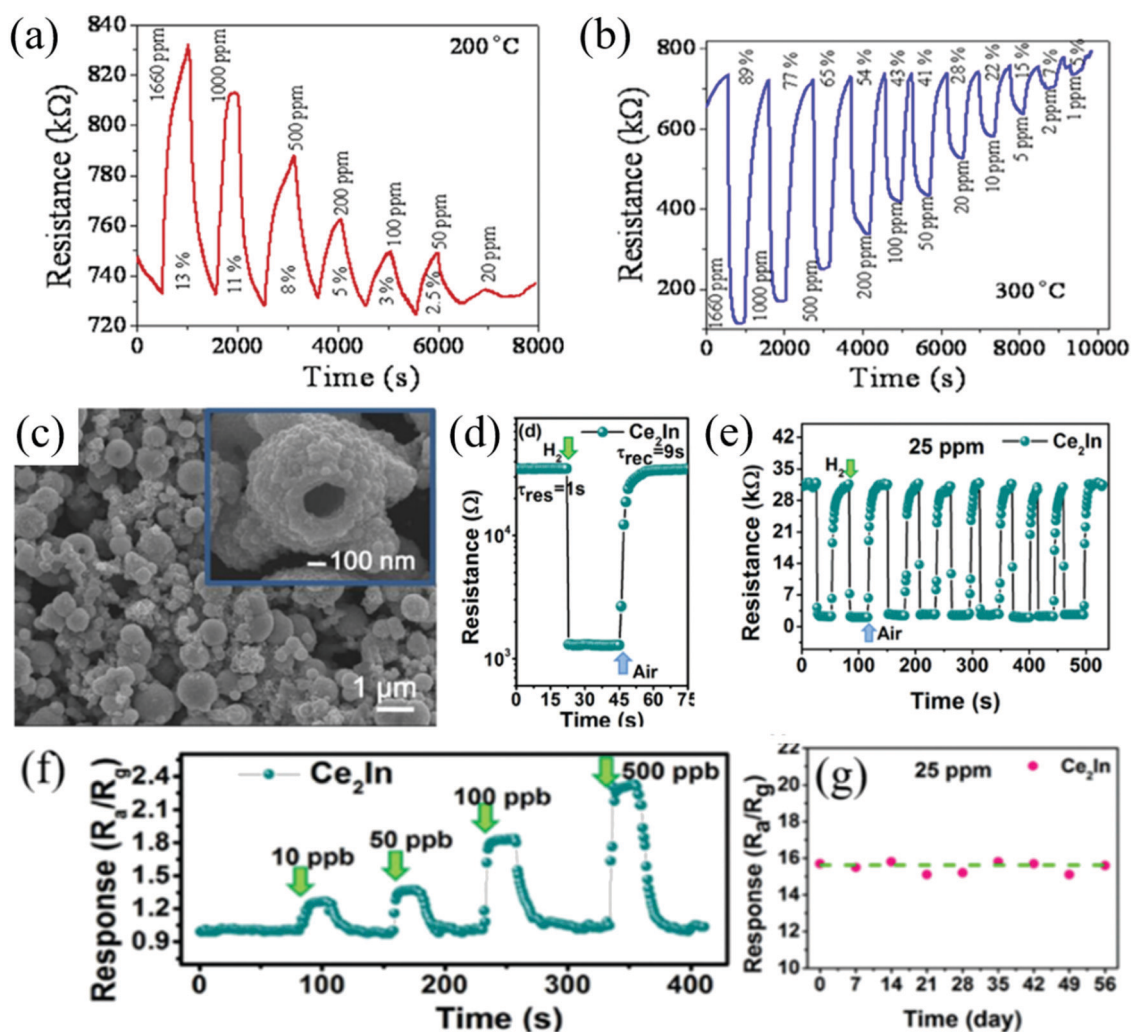


Fig. 15 Resistance transients of indium-doped ZnO thin film gas sensor recorded in the presence of various H₂ gas concentrations: (a) at 300 °C showing n-type behavior and (b) at 200 °C showing p-type behavior²⁶⁸ (Copyright 2014, Elsevier). (c) SEM image of the CeO₂-In₂O₃ hollow sphere; (d) CeO₂-In₂O₃ gas sensor response and recovery curve at 160 °C to 100 ppm H₂; (e) gas sensor reversibility at 160 °C; (f) dynamic response and recovery curve of the prepared gas sensor to ppb level H₂ detection; (g) long-term stability⁸⁵ (Copyright 2018, Elsevier).



again after ageing for two months and the maximum deviation of the sensor response measured during the whole sensing cycle is less than 2% (Fig. 15g). Similarly, other n-n heterostructures for the sensitive sensing of H_2 such as CeO_2 - SnO_2 ,²⁵⁹ ZnO - WO_3 ,²⁶⁴ and Nb_2O_5 - ZnO ²⁶⁵ have also been reported.

(ii) *P-N heterostructure*. The P-N heterostructure is composed of different types of semiconductor materials; thus, the p-n heterostructure has a higher barrier than the n-n heterostructure.²⁷² The p-type semiconductor has a high hole concentration, while the n-type semiconductor has a high electron concentration; at the interface of junction, the electrons and holes recombine at the interface of the two semiconductors, forming a space charge region, leading to an increase of the resistance of the composite in air. It enhances the response capability of the sensors effectively.

Shanmugasundaram *et al.*³⁹ synthesized SnO/SnO_2 nanocomposites in an autoclave using the “one-pot method”. The nanoparticles self-assembled into layered structures similar to nanorods during the reaction and further aggregated to form a loose cubic morphology, which finally transformed into dense microprism structures, each with a particle size of 8–10 nm. The heterogeneous interface between SnO and SnO_2 helps to

accelerate the transfer of free electrons from SnO_2 with high power functions to SnO , thereby improving the sensor response to low concentrations of H_2 . Other p-n heterostructures such as Nb_2O_5 - NiO ²⁹ and Cr_2O_3 - Nb_2O_5 ²⁶⁷ could both realize the detection of low concentrations of H_2 at room temperature.

At present, most of the latest work is focused on improving the sensitivity and response/recovery of the H_2 gas sensor by reducing the grain size, expanding the specific surface area, and improving the gas accessibility. However, improving the selectivity of the H_2 gas sensor is still a major hurdle in this area.

It is found that the selectivity can be regulated by the modulation of the barrier height in the heterojunction through the asymmetric gas sensing reactivity of heteromaterials.^{48,273} Hu *et al.*⁴⁸ prepared the p- Co_3O_4 /n- SnO_2 composite at different Co/Sn molar ratio by the simple soaking calcination method. The composite with different proportions showed peculiar performance. As seen from Fig. 16a, when the molar percentage of Co/Sn was low, the sensor showed n-type sensing response to 50 ppm H_2 ; however, when the molar percentage of Co/Sn was high, the sensor showed p-type sensing response. The sensing mechanism is shown in Fig. 16b and d. When the molar percentage of Co/Sn is low, most of the junctions formed at

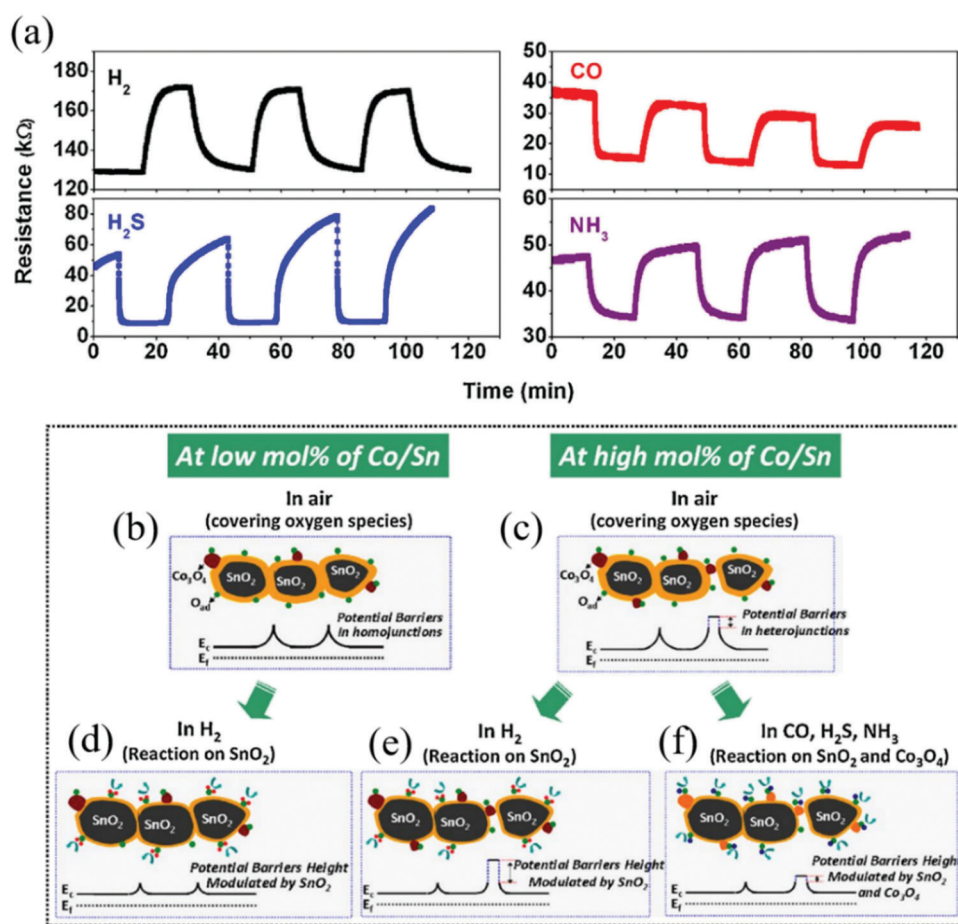


Fig. 16 (a) Co/Sn-10% dynamic resistance curve of typical H_2 , CO, NH_3 , and H_2S reducing gas; (b–f) Co_3O_4/SnO_2 composite sensing mechanism diagram⁴⁸ (Copyright 2017, Elsevier).



the SnO₂/SnO₂ boundary are homojunction. SnO₂ plays a leading role in the resistance modulation of the heterostructures; therefore, the composites show an n-type response to H₂. However, in the case of high molar percentage of Co/Sn, in addition to SnO₂–SnO₂ homojunction, many SnO₂/Co₃O₄ heterojunctions are formed at the same time (mechanism diagram is shown in Fig. 16c, e and f). The formed p–n heterostructure will increase the resistances of the Co₃O₄/SnO₂ composites sharply. After contacting H₂, SnO₂ reacts rapidly with H₂ and electrons are sent back to the conduction band of SnO₂, resulting in a decrease in the bending of the conduction band as a result of the high reactive activity to H₂. On the other hand, Co₃O₄ has little sensitivity to H₂; thus, the conduction band of Co₃O₄ remains almost unchanged. Consequently, the barrier height of the heterostructures is increased instead of decreasing, leading to an increase in the resistance. Unlike H₂, other gases can react with both SnO₂ and Co₃O₄ so that the height of the barrier decreases. The method of modulation of the barrier height in the heterojunction will become an indispensable tool for developing the selectivity of the sensors.

The heterostructure can significantly improve the sensor performance and reduce the detection limit of the sensor due to its extremely high chemical reaction activity as well as the presence of a large number of oxygen vacancies at the interface. Researchers can also achieve excellent selective sensor preparation by adjusting the barrier height of the heterostructures; however, much progress is still needed to reduce the working temperature of the gas sensor.

5.2 MOx–carbon nanocomposites

5.2.1 Synthesis of MOx–carbon nanocomposites. Carbon nanomaterials such as graphene, carbon nanotubes, carbon nanoparticles, and carbon nanofibers have attracted the interest of researchers due to their unique advantages, such as the specifically large surface areas and fast carrier transport. The preparation strategy of MOx–carbon nanocomposites seems easier, usually by multi-step synthesis—simply put, using carbon nanomaterials as substrates and then loading MOx nanomaterials onto the carbon nanomaterials to obtain the composite structures.

Graphene is the most active unit of carbon nanomaterials due to its unique 2D structure, large specific surface area, and excellent electron transport ability, which has attracted extensive attention from researchers.²⁷⁴ The preparation procedure of graphene–MOx nanocomposites in H₂ gas sensing includes two main steps: firstly, the graphene structure is prepared by gas-phase methods such as CVD or liquid-phase Hummers' method, and then the MOx nanomaterials is compounded with it.²⁷⁵

Kamal *et al.*²⁷⁶ employed graphene oxide (GO) and nickel dihydrazine benzoate as a precursor to prepare graphene-loaded nickel oxide (G/NiO) hybrid materials by the chemical method, as described in the flow chart (Fig. 17a). Briefly, at first, the improved “Hummers' method” was used to obtain the graphene structure, then L-ascorbic acid was used to reduce graphene, and finally, nickel dihydrazine benzoate was mixed with graphene in the liquid phase to obtain the composite structures. The complex surface contains a large number of groups, which interact with the surface of graphene, adsorb on the surface of graphene through a covalent bond, and then decompose by the loss of N₂H₄, CO₂, and H₂O through low-temperature heat treatment. Thus, the formed NiO was deposited on G without further purification to obtain G/NiO.

The solvothermal method can obtain MOx nanomaterials with good crystal form and uniform particle distribution; meanwhile, graphene oxide can also be reduced to rGO. Thus, graphene–MOx can be prepared directly. Zhang *et al.*²⁷⁷ synthesized SnO₂ graphene (S-G) composites by a “one-pot” hydrothermal method. Rod-like SnO₂ was grown on the graphene substrate (Fig. 17b and c). Graphene-loaded TiO₂ was prepared by the sol–gel method using titanium tetra-isopropoxide as the precursor.²⁷⁸ Hydroxyl groups on the surface of graphene could also act as nucleation centers of the metal oxide particles, loading more inorganic nanoparticles on the surface of graphene.

Notably, the use of block copolymers can lead to highly regular and periodic structures of materials, which are ideal candidates for the preparation of highly ordered nanoarrays. Liu *et al.*⁸⁹ synthesized C–WO₃ nanoparticles with different morphologies using a co-block polymer compound PS-*b*-P₄VP as the template. The processes are shown in Fig. 17d.

5.2.2 MOx–carbon nanocomposites for H₂ sensing. The carbon-based nanocomposites can improve the sensitivity and

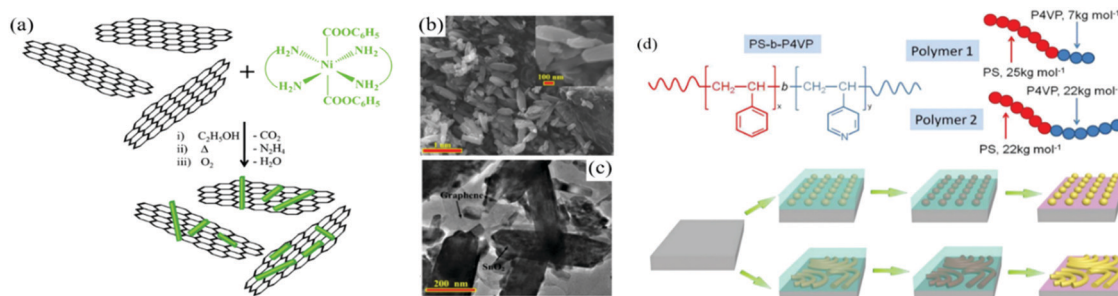


Fig. 17 (a) Synthetic route for the preparation of the G/NiO hybrid²⁷⁶ (Copyright 2016, Elsevier). SEM (b) and TEM (c) images of SnO₂ graphene (S-G) composites²⁷⁷ (Copyright 2016, Elsevier). (d) Structure and molecular weight of the PS-*b*-P₄VP used in this study, and the schematic representation of C-doped WO₃ nanostructure preparation with PS-*b*-P₄VP⁸⁹ (Copyright 2015, Royal Society of Chemistry).



Table 12 Response of MOx-carbon nanocomposites to H₂

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
C-SnO ₂	Spherical	2 vol%	50	—	6% ^d	—	279
C-WO ₃	Nanodots	60 000 ppm	RT	—	114 ^g	162/108 s	89
Graphene-In ₂ O ₃	Nanocrystals	—	250	250 ppm	—	—	90
Graphene-NiO	Thin paper like	2000 ppm	200	—	52.4% ^d	—	276
Graphene-SnO ₂	Flower-like	—	150	—	87.2% ^d	—	277
Graphene-TiO ₂	Thin films	0.5 vol%	125	0.5 vol%	19% ^f	16/61 s	278
Graphene-ZnO	Nanotubes	100 ppm	RT	10 ppm	28.08 ^a	30/38 s	280
Graphene-ZnO	Nanosheet with nanorods	200 ppm	150	10 ppm	3.5 ^a	22/90 s	281
rGO-CuO	Film	1500 ppm	RT	10 ppm	~12 ^d	80/60 s	282
rGO-NiO	Nanosheet	1 vol%	18	0.5 vol%	0.64 ^d	28/142 s	91
rGO-ZnO	Nanofiber	10 ppm	400	100 ppb	2524 ^a	0.9/3.5 min	283
CNT-Co ₃ O ₄	Circle-shaped	—	RT	—	—	21/26 s	88
CNT-SnO ₂	Nanoparticles	4 vol%	200	0.1 vol%	84% ^d	120 s	284
CNT-ZnO	Nanorods	1000 ppm	300	—	66% ^d	—	285
CNF-ZnO	Nanofibers	100 ppm	150	1 ppm	73.54% ^d	29.66/326.22 s	92

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .

selectivity of the H₂ gas sensor, and also reduce the optimal working temperature of the sensor. The details are shown in Table 12. In this section, we will discuss the application of MOx-carbon nanocomposites in H₂ gas sensing, particularly focusing on graphene-MOx nanocomposites.

Since graphene was separated from graphite in 2004, it has attracted extensive attention from researchers because of its huge specific surface area, good stability and conductivity, and high mechanical strength.^{286,287} Pure graphene has poor sensitivity to H₂ and needs to be combined with other materials to make it sensitive.^{274,280} Graphene-MOx nanocomposites have significant advantages in detecting H₂.^{288,289} Firstly, the graphene surface contains a large number of oxygen functional groups, which make it possible to adsorb more H₂, thereby enhancing the resistance modulation ability of the composites. Secondly, the wide specific surface area of graphene and the surface defects existing in the graphene material itself provide huge reactive active sites for the adsorption and reaction of gases and the attachment of MOx nanomaterials on its surface.^{91,290} The presence of these reactive active sites can not only improve the sensor capability of the sensor but also enhance the stability and repeatability of the gas sensor. Finally, graphene has good conductivity and high carrier mobility, which can accelerate the conduction of electrons on the surface and inside the composite materials;^{90,276} also, the improvement of electron conduction efficiency can shorten the response/recovery time of the sensor. These characteristics can reduce the detection temperature, and improve the selectivity and sensitivity.

Ren *et al.*⁹¹ prepared NiO@rGO nanostructures by freeze-drying combined with heat treatment. The SEM and TEM images of the NiO@rGO nanocomposites are given in Fig. 18a and b. It can be found that there are some wrinkles on the nanocomposites and a large number of NiO nanoparticles are uniformly distributed on the surface of rGO. The material-modified sensor has excellent performance with a relative response of 0.64 (defined as $\Delta R/R_a$) to 1 vol% H₂. This is because both NiO and graphene are p-type semiconductor materials; thus, p-p heterojunctions are generated on the

surface of the NiO@rGO nanocomposites, which can increase the number of holes and its macro resistance. When exposed in H₂ atmosphere, reactive oxygen anions produced by heterojunctions will react with H₂ rapidly; coupled with the internal high load of graphene, the electrons will be immediately released to the metal surface, which will greatly increase the ability of resistance modulation, thus achieving a rapid, efficient, and sensitive detection of H₂.

The performance of the sensor is shown in Fig. 18c and d. Kamal *et al.*²⁷⁶ also studied the H₂ gas sensing properties of the NiO/rGO nanostructures. They found that the high porosity of graphene and a large number of single active sites on graphene are the key factors to improve the performance of the sensor. Moreover, due to the strong electronic transmission ability of graphene, the close contact between NiO and graphene will lead to electronic interaction between the two materials, which may lead to the enhanced charge separation of the composite material, thereby enhancing the performance of the sensor.

Compared with the composite of p-type semiconductor and graphene, the composite of n-type semiconductor nanostructure and graphene is more widely used in gas sensing because the p-n junction can be formed.

For example, the GO/ZnO composite prepared by Anand *et al.*²⁸¹ achieved a reduction in the sensing temperature and the sensor exhibited higher performance at lower temperature compared with the corresponding bare ZnO. It is because graphene interacts weakly with the ZnO nanostructures through weak van der Waals force; thus, it can maintain its original high carrier mobility in the GO/ZnO composites. Lamellar graphene as a substrate provides a wide 3D network for the composite materials, which makes interconnection between the ZnO nanorods closer. The surface defects and functional groups of graphene themselves can also serve as high-energy adsorption sites for gas molecules. At lower temperatures, H₂ can realize the adsorption, reaction, and desorption processes on the surface of the material, thus reducing the detection temperature and realizing low power detection. The sensor can realize a detection limit of 10 ppm at a low temperature.



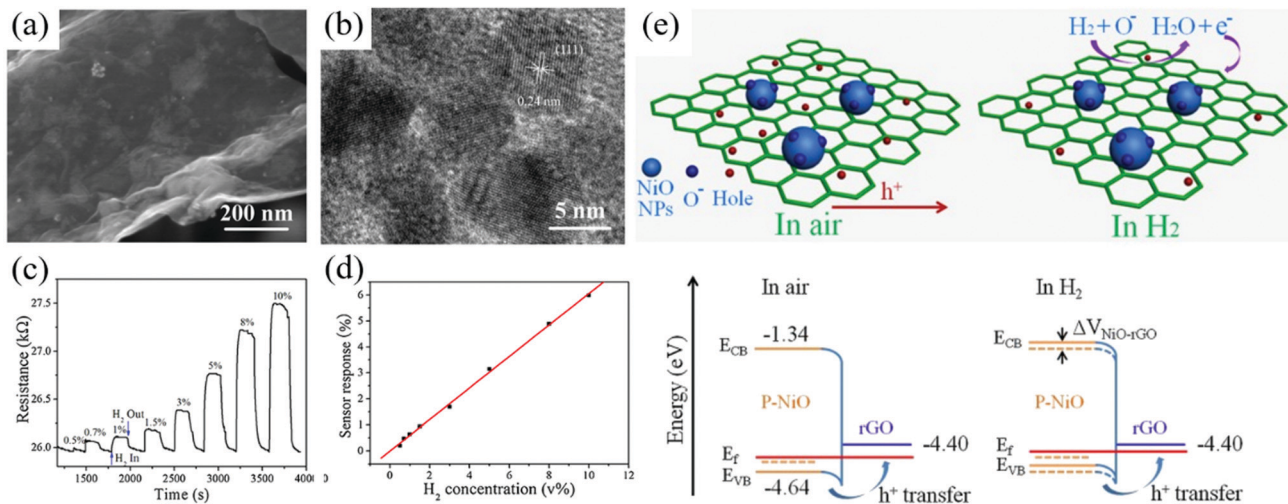


Fig. 18 (a) SEM image; (b) HRTEM image of the NiO@rGO nanocomposite; (c) the NiO@rGO sensor operating at a temperature of 50 °C and RH 40%; the response curve of the corresponding response when exposed to various H₂ concentrations under the condition of 18 °C; (d) linear fitting diagram; (e) schematic diagram of the possible sensing mechanism of the NiO@rGO nanocomposite exposed to H₂ gas, energy band diagram of NiO@rGO, and the hole transfer between NiO and rGO⁹¹ (Copyright 2018, Elsevier).

There are also some H₂ gas sensors based on graphene-based composite nanostructures such as p-TiO₂/n-graphene²⁷⁸ and (rGO)-CuO²⁸² sandwich structures, which also achieve low power detection of H₂ due to the synergistic effect of metal oxide nanoparticles and the graphene structure.

In addition to lowering the working temperature of the sensor, graphene has also contributed significantly to the improvement of the performance of the H₂ sensor. Kathiravan *et al.*²⁸⁰ prepared a novel ZNT/G nanohybrid structure by combining the hydrothermal route with CVD. At room temperature, ZNT/G shows excellent performance to H₂ at low concentration and has good repeatability, strong stability, as well as ultra-fast response and recovery to H₂. Mansha *et al.*⁹⁰ also achieved the enhancement of the H₂ sensing performance by the hydrothermal synthesis of the In₂O₃/graphene heterostructure.

Other carbon materials such as carbon nanoparticles, carbon nanotubes, and other composites also have a strong sensing performance after combining with MOx. It has been reported that the narrower the bandgap of the C nanoparticle-doped metal oxides, the lower the energy required for the transition of electrons from the valence state to the conduction band; thus, it is easier to realize gas sensing detection at a low power. The hexagonal C-WO₃ nanorod arrays prepared by Liu *et al.*⁸⁹ have a unique performance as the nanoarrays were interconnected by ultrathin carbon films because of a new conductive path. When H₂ is encountered, the conductivity modification of each C-doped WO₃ nanoparticle will be rapidly transferred through this new conduction pathway, thus achieving an improved hydrogen response. Also, because the work function of WO₃ (4.30 eV) is lower than that of C (4.81 eV), at the heterogeneous interface, electrons are transferred from the C-doped WO₃ nanoparticles to the carbon film so that they accumulate at the carbon film. When H₂ enters, the resistance changes dramatically, which enhances the performance of H₂ sensing. Bhatnagar *et al.*²⁷⁹

studied the behavior of C-doped SnO₂ nanoparticles in gas sensing. They found that the material exhibited p-type sensing behavior for H₂, n-type for ethanol, and achieved highly selective detection for H₂.

Other carbon nanomaterials with different morphologies, such as carbon nanofibers⁹² and carbon nanotubes²⁸⁵ could pose high sensitivity to low concentration H₂ after combination with MOx nanomaterials.

In conclusion, the performance of the sensor can be enhanced effectively by combining carbon nanomaterials. The surface defects of graphene itself greatly increase the active adsorption sites of the gases and the existence of heterostructures after compounding with MOx nanomaterials effectively improves the electron transport kinetics. The carbon nanotube structure can attach more active particles due to its porous structures, which enhances the contact with the target gas. Carbon particle doping can sensitize the sensor by changing the band structure of the MOx nanomaterials.

6. Ternary or more complicated nanostructures for H₂ gas sensors

There is much potential for the utilization of ternary or more complicated nanostructures as sensing materials due to the more synergistic effects among the different elements in the nanocomposite. A few quintessential synthetic strategies and the application in H₂ sensing are given in this section, and at the end of this section, some flexible H₂ gas sensors based on polymer/MOx are included.

6.1 Synthetic strategy for ternary or more complicated nanostructures

A few methods for the preparation and tailoring of ternary or more complicated nanostructures have been reported in the



Table 13 Details of the synthesis of ternary or more complicated nanostructures

Materials	Method	Solvent	Temp., time	Conv. cond.	Ref.
CoO/Co ₃ O ₄ /WO ₃	One-step hydrothermal	H ₂ O	180 °C, 24 h	600 °C, 6 h	291
Pd–SnO ₂ –MoS ₂	Multistep hydrothermal	H ₂ O	200 °C, 24 h; 120 °C, 12 h; 180 °C, 16 h	700 °C, 2 h	94
Pt–TiO ₂ –MoS ₂	Multistep hydrothermal	H ₂ O	180 °C, 24 h; 200 °C, 24 h	400 °C, 1 h	292
Pd/WO ₃ –ZnO	Reactive DC magnetron sputtering technique	Ar:O ₂ (4:1)	WO ₃ : 60 W, ZnO: 50 W, Pd: 30 W	—	293
rGO–Ni–ZnO	Ni–ZnO: RF sputtering techniques Hummers' method	rGO: —	50 W, 25 sccm, 600 °C 2 h, 1.5 × 10 ⁻² mbar	—	96
Pd–Al ₂ O ₃ –TiO ₂	Al ₂ O ₃ –TiO ₂ : ALD Pd: e-beam evaporation	Carrier gas: Ar	—	—	93
Pd/TiO ₂ /ITO	TiO ₂ /ITO: spray deposition method Pd/TiO ₂ /ITO: electron beam deposition	—	Deposition pressure: 10–5 mTorr	300 °C, 2 h	294
Pd–ZnO–BN	ZnO: vapor growth Pd BN: ALD	Vapor growth: Ar and O ₂ ALD: Ar	ZnO: 950 °C, 1 h BN: 750 °C, 40 cycles Pd: 220 °C, 200 cycles	—	105

past. In the preparation process of materials, it is more necessary to select the combination of different methods to achieve the morphological control of composites and the precise control of the stoichiometric ratio of the components. The synthetic strategies are summarized in Table 13.

The hydrothermal method still withstands the precise control of each component in multicomponent materials. In this regard, the size and morphology of each component can be effectively controlled, and the product has good dispersion. For example, Luo *et al.*²⁹² synthesized the Pt–TiO₂–MoS₂ composite nanostructures by a two-step hydrothermal method and then annealed it by blending with Pt precursors to obtain ternary composite structures. In short, TiO₂ was synthesized using HF and TBOT as the precursors in a typical hydrothermal way. Then, the synthesized TiO₂ NPs were used as raw materials, mixed with sodium molybdate dihydrate and thioacetamide in solution, and the TiO₂–MoS₂ composite was again obtained hydrothermally. To generate such a composite is relatively simple. Mixing TiO₂–MoS₂ and chloroplatinic acid in absolute ethanol, a uniform paste is obtained after ultrasonication and annealing at 400 °C; then, a ternary composite structure could be obtained. Zhang *et al.*⁹⁴ also used a multistep hydrothermal method to obtain the Pd–SnO₂/MoS₂ composites.

The synthesis of multiple composites doped with noble metal nanoparticles includes mainly two steps: the synthesis of MOx nanomaterials by liquid-phase methods and then the decoration of noble metal nanoparticles onto the surface of MOx nanomaterials by vapor deposition. Hassan *et al.*⁹⁵ prepared bi-noble metal-doped ternary composites, Pt/Pd–ZnO, using a combination of liquid and gas-phase synthesis methods. In a nutshell, vertically aligned ZnO nanorods were obtained by the template-assisted method and the Pt/Pd structure was synthesized by the PLD technology with high purity Pt and Pd as the targets.

6.2 Ternary or more complicated nanostructures for H₂ sensing

Ternary or more complicated nanostructures are considered to be better for detecting and monitoring H₂ as compared to others because they can introduce more active sites, which are beneficial for gas sensing sensitization into the material to

amplify the gas sensing signal and improve the selectivity, stability, and repeatability. The characteristics and properties of different kinds of these composite materials to H₂ are summarized and listed in Table 14.

Compared with a single noble metal, bi-noble metal nanoparticles exhibit different adsorption and catalytic properties due to its geometrical and electronic effects. For example, Fan *et al.*²⁹⁸ used bi-noble metal-doped Pt–Au@ZnO ternary nanostructures as sensing materials for H₂ detection. The morphology of Pt–Au@ZnO is shown in Fig. 19a and b. The sensor can respond to H₂ at the ppm level at room temperature. In terms of the structure, after doping bi-noble metals, isolated Au and Pt nanoparticles will be formed on the surface of the composite, resulting in highly active sites, thus increasing the adsorption and catalytic activity of bi-noble metal nanoparticles that enable the strong adsorption of H₂ on the surface of the composite. Regarding electron transport, the presence of bi-noble metal nanoparticles will accelerate the electron transport rate and reduce the sensing response time. The response range of the Pt–Au@ZnO nanocomposites to H₂ is relatively high. As shown in Fig. 19c, this material can detect H₂ in the concentration range of 50–2000 ppm and the response signal has a good linear relationship. Moreover, the sensor has a good selectivity (Fig. 19d).

To further improve the sensing performance of the sensor, Hassan *et al.*⁹⁵ prepared Pt–Pd nanomaterials with a core–shell structure. The selectivity of the sensor is also greatly enhanced due to the presence of the noble metal compared with the bulk materials. The core–shell structures of bi-noble metals have shown different adsorption characteristics for H₂ due to their different structures and electronic states. As the Pt–Pd core–shell structure is uniformly covered with a thin layer on the surface of the ZnO nanorods (Fig. 19e and f), when the sensor is exposed to the H₂ atmosphere, H₂ can easily penetrate through the Pt–Pd structure to reach the surface of the ZnO nanostructures. The sensing reaction occurs at the interface and the signal is increased.

Due to the high hydrophobicity and low surface energy on the surface of the material, the grain size and the grain spacing become smaller, which leads to particle contact between



Table 14 Details of the H₂ gas sensor based on ternary or more complicated nanostructures

Materials	Structure	<i>c</i>	<i>T</i> (°C)	LOD	Sensitivity	Response time	Ref.
Pd–Al ₂ O ₃ –TiO ₂	Thin film	5 ppm	450	5 ppm	15% ^f	—	93
Pd–In ₂ O ₃ –ZnO	Nanofibers	50 ppb	350	0.05 ppm	172 ^a	—	295
Pd–SnO ₂ –MoS ₂	Nanosheets	30 ppm	RT	30 ppm	0.4% ^d	5 s	94
Pd–TiO ₂ –ITO	Films	100 ppm	RT	25 ppm	35 ^e	15 s	294
Pd–WO ₃ –ZnO	Thin films	100 ppm	200	10 ppm	16.8 ^a	16/62 s	293
Pd–Zn ₂ SnO ₄ /ZnO	Nanowires	1000 ppm	400	—	15.6 ^a	2–4 min/5 min	296
Pd–ZnO–BN	Nanowires	10 ppm	200	0.5 ppm	12.28 ± 0.61 ^a	160/90 s	105
Pd–ZnO–NiO	ZnO: nanorods NiO: nanoplates	100 ppm	225	2 ppm	72% ^d	—	118
Pt–Pd–ZnO	Core-shell nanorod clusters	10 000 ppm	100	0.2 ppm	58% ^d	5 s	95
Pt–Pd–ZnO	Nanorods	10 000 ppm	100	0.2 ppm	69.8 ^d	5/76 s	297
Pt–TiO ₂ –MoS ₂	Flower-like	500 ppm	100	50 ppm	47.09% ^d	2.5/6.17 min	292
Pt–Au–ZnO	Nanorods	250 ppm	RT	50 ppm	25 ^a	115 s	298
Pt/F-MWCNTs–SnO ₂	Nanotubes	0.05 vol%	RT	0.05 vol%	5.4% ^d	300 (1 vol% H ₂)	299
Pt/F-MWCNTs/TiO ₂	—	0.05 vol%	25	—	3.8% ^d	—/20 s	97
rGO–Ni–ZnO	—	100 ppm	150	1 ppm	63.8% ^d	28 s	96
rGO/Au/ZnO	—	500 ppm	RT	—	96 ^d	8/612 s	300
rGO/Nb–TiO ₂	—	120 ppm	200	—	28.4 ^d	930/≅300 s	301
rGO–Ni–ZnO	rGO: nanosheet Ni–ZnO: nanowires	50 ppm	150	1 ppm	29.9% ^d	70/180 s	302
rGO–Pd–ZnFe ₂ O ₄	Nanoparticles	1000 ppm	RT	50 ppm	29.7% ^d	18/39 s	303
UNCD–MoS ₂ –ZNRs	Nanorod	100 ppm	RT	50 ppm	50.3 ^a	8/12 s	304
PAN–TiO ₂ –SnO ₂	Spherical	0.8 vol%	27	—	6.18 ^a	245/57 s	296
Pt–IGZO	Thin films	0.01 vol%	RT	100 ppm	100 ^f	~300 ms/30 s	305

LOD: limit of detection; response with different definition *a*: R_a/R_g , *b*: R_g/R_a , *c*: I_a/I_g , *d*: $\Delta R/R_a$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *e*: $\Delta R/R_g$, $\Delta R = (R_a - R_g)$ or $(R_g - R_a)$, *f*: $\Delta I/I_a$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *g*: $\Delta I/I_g$, $\Delta I = (I_a - I_g)$ or $(I_g - I_a)$, *h*: I_g/I_a .

adjacent particles in the H₂ atmosphere, thus increasing the number of shortest current paths, resulting in shorter sensor response/recovery time (Fig. 19g) and the sensor can realize the response to H₂ under high humidity conditions (Fig. 19h). Meanwhile, the sensor also has excellent selectivity for H₂ (Fig. 19i).

Besides, bi-noble metal modified MOx nanocomposites can also be used for the preparation of flexible devices,²⁹⁷ which provides theoretical support for the application of wearable devices. In addition to bi-noble metal doping, some bimetal-doped metal oxide composite structures such as Pt/Nb–TiO₂/Pt³⁰⁶ and Ga–Pd/ZnO²⁴⁰ also have a high response to H₂.

Because of a large specific surface area and special physico-chemical properties, 2D materials are often used in multicomponent composite structures to “connect” other components or to load multicomponent substances. Among them, graphene is still widely used in ternary composites due to the wide specific surface area and abundant surface defects. Bhati *et al.*⁹⁶ prepared rGO–Ni–ZnO composites by RF sputtering. rGO is a p-type semiconductor and can form various heterostructures with Ni-doped ZnO. The intrinsic properties of rGO improve the active sites for oxygen atom adsorption and increase the loading of oxygen on the material surface.

As mentioned above, rGO extracts electrons from ZnO, which will increase the thickness of the depletion layer and increase the material to macro. The modulation ability of the apparent resistance enables the efficient and sensitive detection of H₂. Abdollahi *et al.*³⁰² also synthesized the rGO–Ni–ZnO ternary composite by the hydrothermal method, which also has high sensitivity and selectivity for H₂. At the working temperature of 150 °C, the sensor response value to 50 ppm of H₂ is

29.9% (refined as $\Delta R/R_a$). The role of graphene in ternary composites is to serve as the substrate of the sensing materials, which could provide reactive active sites, improve the resistance modulation ability of the composites, and thus improve the performance of the sensors.

MoS₂, another 2D layered lamellar materials with graphene shape, is considered as a promising gas sensing material due to its specific surface area.⁷⁰ MoS₂ has good electrical properties. Unlike graphene, MoS₂ has a bandgap ranging from 1.2 to 1.8 eV due to different microstructures, which provides a broad space for researchers to regulate it. MoS₂ with a wide specific surface area as a substrate can act as a carrier for the material and gas adsorption, and can bring more surface-active sites so that the sensor can obtain better performance. At the same time, the heterostructures generated during the composite with metal oxides also contribute to the improvement of the sensor performance.

Zhang *et al.*⁹⁴ prepared the Pd–SnO₂/MoS₂ ternary hybrid for H₂ gas sensing. Due to the synergistic effect of ternary nanostructures and the regulation of the barrier by electron transport, the sensor can detect H₂ at room temperature with exceedingly good performance. The characterization test analysis showed that the Pd–SnO₂/MoS₂ composite was a mesoporous material and its BET surface area was calculated to be 108.93 m² g^{−1}. According to the Barrett–Joyner–Halenda (BJH) algorithm, the pore size distribution of the Pd–SnO₂/MoS₂ composite is about 5.36 nm.

Moreover, the presence of Pd nanoparticles can dissociate H₂ into H atoms on the Pd surface and further accelerate the transfer of free electrons from Pd to the SnO₂/MoS₂ heterogeneous interface, which will enhance the response of the sensor (Fig. 20a and b). Due to the difference in the material



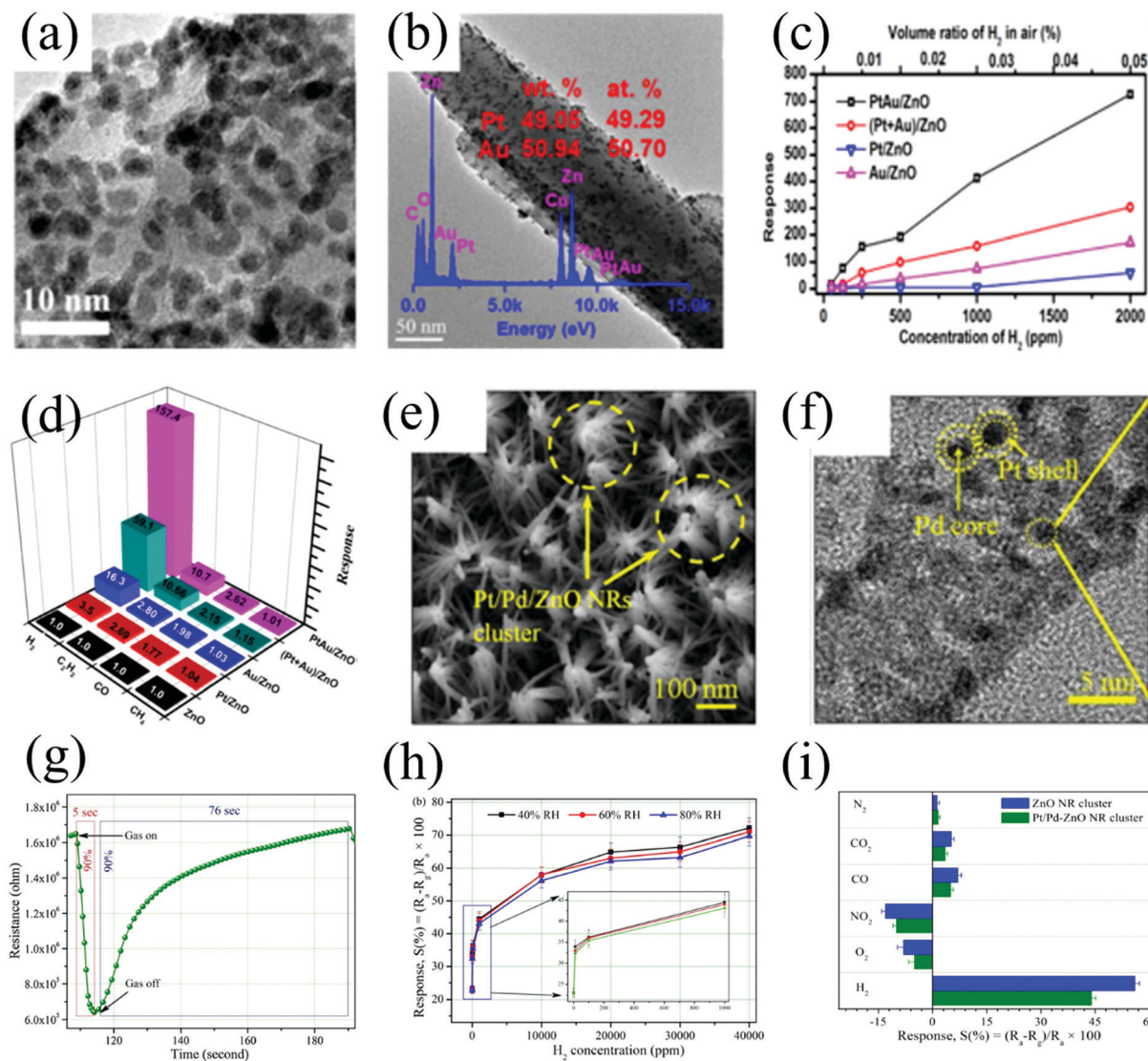


Fig. 19 (a and b) ZnO nanorods loaded with Pt–Au nanoparticles; (c) sensor response to different concentrations of H₂; (d) the gas sensing response of samples to different gases at 130 °C²⁹⁸ (Copyright 2017, Elsevier). (e) SEM image of the Pt/Pd–ZnO NR cluster; (f) TEM image of the Pt/Pd–ZnO core-shell structure; (g) Pt/Pd–response/recovery time characteristic diagram of ZnO NRs/Si exposed to 10 000 ppm H₂ at 100 °C; (h) transient response of the as-fabricated sensor under different humidity at 100 °C. (inset) Enlargement of the curve within the 0.2–1000 ppm range of H₂ concentration; (i) histogram of selectivity of the ZnO NR clusters and Pt/Pd–ZnO NRs clusters to various test gases⁹⁵ (Copyright 2017, Elsevier).

bandgap, Schottky potential barrier is formed between the material and the material interface, and the ability of the composite material to adjust the macroscopic resistance is improved, thus improving the performance of the sensor (Fig. 20c), which makes the ternary composites have a wide application prospect. Pt–TiO₂–MoS₂²⁹² could also respond to H₂, which was obtained by the hydrothermal method. Because of the large surface-to-volume ratio, MoS₂ can improve the good adsorption sites and provide an excellent sensing platform for H₂ detection. The use of MoS₂ also has a certain contribution to improving the stability of the sensors. The UNCD/MoS₂/ZnO nanorod structure shows excellent stability during 60 days' testing.³⁰⁴

In addition to MoS₂, BN has a layered structure similar to graphite and their atomic layers are stacked in different ways. BN is highly insulated; different morphologies have wide

bandgaps from 5 eV to 6 eV.¹⁰⁴ Compared with carbon materials such as graphene, BN has high thermal and chemical stability, excellent thermal conductivity, super mechanical strength, and high oxidation resistance.³⁰⁷ BN has a great potential in gas adsorption and the insertion of the nitride layer into the metal/semiconductor interface has been found to cause great changes in the hydrogen sensing performance, which contributes to the efficient selection of sensors for H₂.

Weber *et al.*¹⁰⁵ prepared a ternary nanowire structure of ZnO–BN–Pd. The detection limit of this sensing material for H₂ was 0.5 ppm. It is the existence of BN that optimizes the selectivity of the sensors. Small voids on the surface of BN can make a highly sensitive choice for the gas. Because the dynamic diameter of the H₂ molecule is small, it can penetrate the nitride layer completely, while it is not easy for the larger



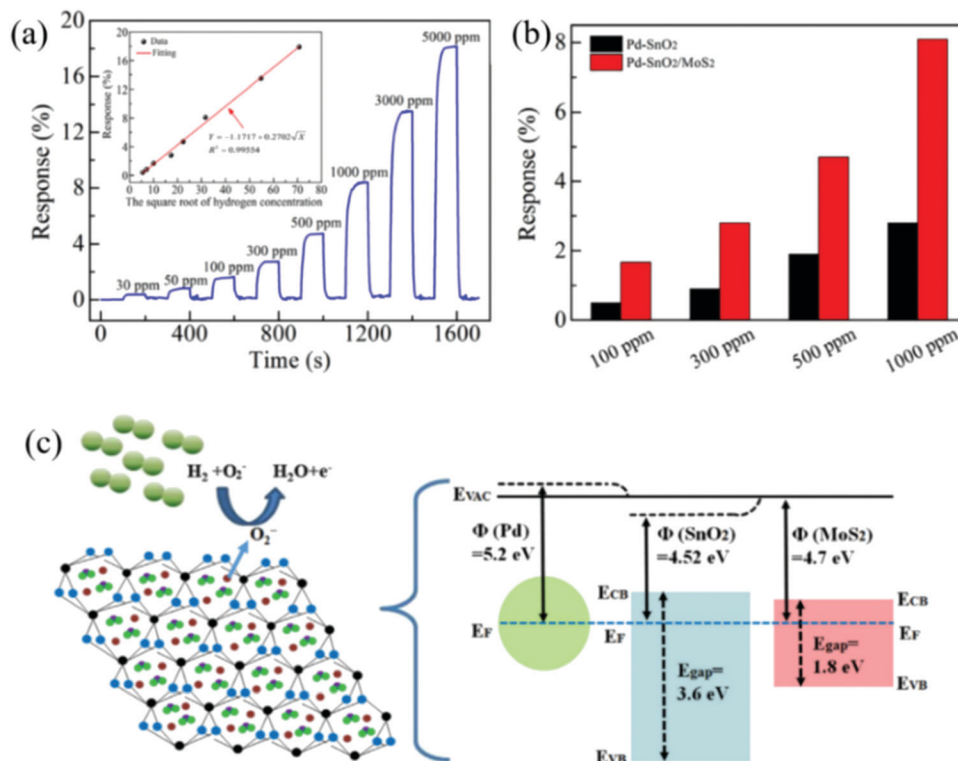


Fig. 20 (a) Response of the Pd–SnO₂/MoS₂ film sensor towards hydrogen in the range of 30–5000 ppm (inset indicates the fitting of normalized response Y as a function of hydrogen concentration); (b) response of the Pd–SnO₂/MoS₂ and Pd–SnO₂ sensors (4 wt% Pd loading) toward 100, 300, 500, and 1000 ppm hydrogen at room temperature; (c) hydrogen-sensing mechanism for the Pd–SnO₂/MoS₂ film⁹⁴ (Copyright 2017, Elsevier).

reducing gas molecule. Through the nitriding layer, it is equivalent to achieving the sensor's choice of H₂.

Other ternary materials^{93,105,293–295,299} are also mostly presented by the combination of some sensitizing structures or substances. By adjusting the electronic transmission and macroscopic resistance, the sensor response capacity can be improved and the selectivity can be optimized, along with the enhancement of the repeatability and stability.

6.3 Flexible H₂ gas sensors based on polymer/MOx

With the development of social economy and the demands of people, flexible H₂ gas sensors based on conducting polymers such as polyaniline (PANI), polypyrrole (PPy), and polyvinylidene fluoride (PVDF) have drawn attention because of their distinctive conductivity, strong mechanical strength, and high operability.^{30,308}

Polymers have been considered as promising candidates for flexible devices and wearable electronics.³⁰⁹ In particular, when combined with MOx, the rigidity of MOx can be improved; meanwhile, the polymer/MOx organic–inorganic hybrid composites will be formed, which exhibit excellent gas sensing performance, including flexibility and room temperature detection.

Among multitudinous polymers, PANI is considered as a potential candidate as a H₂ sensing material due to its environmental stability, controllable electrical conductivity, ease of synthesis, as well as interesting redox properties.³¹⁰

For example, Neri *et al.*³¹¹ synthesized polyaniline/Sn₂O₃ (PANI/Sn₂O₃) nanocomposite with 5 wt% Sn₂O₃ and realized

H₂ detection at room temperature (30 °C) with fast response/recovery time and excellent response repeatability. The excellent performance is attributed to the formation of the p–p heterojunctions between PANI and Sm₂O₃. Similarly, Moghaddam *et al.*³¹² synthesized a series of PANI–MOx organic–inorganic hybrid composites and investigated their H₂ sensing performance. In addition, some other polymer/MOx composites could also realize the highly sensitive detection of H₂ at room temperature.³¹³

In summary, flexible devices as a novel gas sensing platform have been sought after by researchers in recent years.³¹⁴ The combination between polymers and MOx provides sensors for both flexibility and robustness. Moreover, the hybridization of organic and inorganic materials also enhances the performance of the sensors at room temperature. However, compared with other gas sensors, there is less research on flexible H₂ gas sensors. Meanwhile, flexible H₂ gas sensors with excellent sensing response, reliability, and stability are supposed to be further studied.

7. Conclusions and outlook

7.1 Conclusions

This review summarizes and discusses the recently advanced developments in designing MOx-based materials for H₂ gas sensing, the corresponding mechanisms, as well as performance, focusing mainly on the following aspects.



(1) It is crucial for the controllable synthesis of MOx semiconductor materials for gas sensing by selecting appropriate methods and strategies. The target products can be obtained by controlling and optimizing the pertinent reaction variables, such as reaction temperature, concentrations, molar ratios, as well as surface coating molecules.

(2) Regarding the mono-MOx-based sensing materials, the improvement of H₂ sensing response can be achieved by optimizing their morphologies with large surface area and electronic structures, as well as surface defects.

(3) Heteroatom modification is a useful method to boost the performance of the sensors, whether it is noble metal decoration or other metal doping. The surface-decorated noble metal particles, forming Schottky barrier with MOx, can effectively lower the working temperature of the sensor materials, enhance the response ability, reduce the response recovery time, and improve the selectivity of the H₂ gas sensor simultaneously. Moreover, the doping of other kinds of metal nanoparticles may lead to the lattice distortion of MOx as well as increase the surface defects, oxygen vacancies, and surface-active sites, which may greatly improve the resistance modulation ability of the sensing materials.

(4) Binary metal oxide nanostructures are promising for H₂ sensing. Heterogeneous interfaces between the two kinds of materials are the most chemically active regions, in which there are a large number of defects, active sites, and oxygen functional groups. These binary metal oxide nanostructures can improve the performance of the sensors effectively.

(5) Ternary or more complicated nanostructures could provide more effective components to improve the sensor performance and the combination of different structural units will lead to synergistic effects, improving the sensor's all-round.

7.2 Outlook

Advanced developments have been made in H₂ gas sensors based on MOx nanomaterials with the continuous innovation of material synthesis methods and the continuous development of gas sensing platform construction. However, the existing problems in sensing H₂ cannot be ignored; thus, there is still much room for development in the future, as demonstrated below.

Firstly, the performance of H₂ sensors should be further improved. One of the major challenges is to improve the selectivity, which is crucial for H₂ detection. Then, the fast response speed is also vital because of the explosive nature of H₂. Moreover, the high working temperature (*e.g.*, >250 °C) still hinders the commercialization of H₂ gas sensors, which brings inconvenience to its practical application. Therefore, to solve the obstacles, it is of great significance to develop new functional nanomaterials for H₂ gas sensors. Co-doping MOx maybe one of the promising methods because of the synergistic effects of different dopants, which could enhance the performance of the sensors. Third, the development of inorganic-organic hybrid materials is also conducive for the improvement of the H₂ sensing properties.

Although some gas sensing mechanisms have been proposed, there are still many unresolved problems. To better

understand the fundamentals, the use of *in situ* characterization technology will be a powerful tool. It could reveal the process of H₂ adsorption and desorption as well as the reaction that occurs at the surface. Furthermore, computational methods such as density functional theory (DFT) and molecular dynamics (MD) simulations could also help researchers to deepen their understanding of the mechanism of gas sensing.

At present, most H₂ gas sensors have also stayed at the laboratory stage; laboratory tests contain standard errors of instrumentation. Thus, further studies are needed for improving the trueness and stability of the data during long-term outdoor work of the sensor. In addition, in order to extend the usage scenarios of H₂ sensors, the development of visualization of sensing signals and new data analysis strategies are much needed. The change from stoichiometric signals to more widely-accepted prion signals is also a trend in future sensor functionalization.

With the rapid development of the Internet of Things, portable gas sensing devices are highly needed. The integration of sensors can be achieved through the Internet of Things, and the detected information can be collected and shared rapidly, providing users with better service and giving researchers more big data support. In a highly integrated platform, however, there is still a long way to for miniaturizing and integrating gas sensors, and applying them to daily life. Importantly, reducing the cost of sensing materials, simplifying the sensor construction process, and achieving green, simple, and safe preparation will be further studied.

Moreover, flexible devices for detecting H₂ gas have become more attractive. Flexible devices have greater flexibility and ductility, which is also available to be combined with smart glass and designed as wearable electronics. The flexible H₂ gas sensor is attached to the surface of smart windows to realize the real-time monitoring of H₂ in a specific space. For example, the wearable devices could provide timely feedback of our physical health conditions by detecting the concentration of H₂ in our breath.

To sum up, this review aims to provide valuable information to researchers and constructive guide for the future development of H₂ sensors. It is hoped that with interdisciplinary development, more new vitality can be injected into this field, ultimately achieving the upgradation of gas sensors from traditional devices to integrated chip devices, for maximizing the performance.

Conflicts of interest

The authors declare no conflict of interest.

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References

- 1 S. Han, Q. Yun, S. Tu, L. Zhu, W. Cao and Q. Lu, *J. Mater. Chem. A*, 2019, **7**, 24691–24714.
- 2 J. Zhu, L. Hu, P. Zhao, L. Y. S. Lee and K.-Y. Wong, *Chem. Rev.*, 2020, **120**, 851–918.
- 3 C. Wadell, F. A. A. Nugroho, E. Lidström, B. Iandolo, J. B. Wagner and C. Langhammer, *Nano Lett.*, 2015, **15**, 3563–3570.
- 4 E. Menumerov, B. A. Marks, D. A. Dikin, F. X. Lee, R. D. Winslow, S. Guru, D. Sil, E. Borguet, P. Hutapea, R. A. Hughes and S. Neretina, *ACS Sens.*, 2016, **1**, 73–80.
- 5 G. Rahamim, M. Mirilashvili, P. Nanikashvili, E. Greenberg, H. Shpaisman, D. Grinstein, S. Welner and D. Zitoun, *Sens. Actuators, B*, 2020, **310**, 127845.
- 6 K. Nguyen, C. M. Hung, T. M. Ngoc, D. T. Thanh Le, D. H. Nguyen, D. Nguyen Van and H. Nguyen Van, *Sens. Actuators, B*, 2017, **253**, 156–163.
- 7 Q. A. Drmosh and Z. H. Yamani, *Ceram. Int.*, 2016, **42**, 12378–12384.
- 8 B. R. Huang, J. P. Chu, A. Saravanan, M. M. Yenesew, N. Bönninghoff and C. H. J. C. A. E. J. Chang, *Chem. – Eur. J.*, 2019, **25**, 10385–10393.
- 9 D. Sil, J. Hines, U. Udeoyo and E. Borguet, *ACS Appl. Mater. Interfaces*, 2015, **7**, 5709–5714.
- 10 M. Cho, J. Zhu, H. Kim, K. Kang and I. Park, *ACS Appl. Mater. Interfaces*, 2019, **11**, 13343–13349.
- 11 V. Krivetskiy, A. Efitorov, A. Arkhipenko, S. Vladimirova, M. Rumyantseva, S. Dolenko and A. Gaskov, *Sens. Actuators, B*, 2018, **254**, 502–513.
- 12 J.-H. Kim, A. Mirzaei, H. Woo Kim, P. Wu and S. S. Kim, *Sens. Actuators, B*, 2019, **293**, 210–223.
- 13 A. Dey, *Mater. Sci. Eng., C*, 2018, **229**, 206–217.
- 14 C. Zhang, G. Liu, X. Geng, K. Wu and M. Debligny, *Sens. Actuators, A*, 2020, **309**, 112026.
- 15 S. Roso, C. Bittencourt, P. Umek, O. González, F. Güell, A. Urakawa and E. Llobet, *J. Mater. Chem. C*, 2016, **4**, 9418–9427.
- 16 C.-C. Chueh, C.-I. Chen, Y.-A. Su, H. Konnerth, Y.-J. Gu, C.-W. Kung and K. C. W. Wu, *J. Mater. Chem. A*, 2019, **7**, 17079–17095.
- 17 C. C. Lee, C. I. Chen, Y. T. Liao, K. C. W. Wu and C. C. J. A. S. Chueh, *Adv. Sci.*, 2019, **6**, 1801715.
- 18 H. Konnerth, B. M. Matsagar, S. S. Chen, M. H. G. Prechtel, F.-K. Shieh and K. C. W. Wu, *Coord. Chem. Rev.*, 2020, **416**, 213319.
- 19 E. Doustkhah, J. Lin, S. Rostamnia, C. Len, R. Luque, X. Luo, Y. Bando, K. C.-W. Wu, J. Kim, Y. Yamauchi and Y. Ide, *Chem. – Eur. J.*, 2019, **25**, 1614–1635.
- 20 Y.-T. Liao, B. M. Matsagar and K. C. W. Wu, *ACS Sustainable Chem. Eng.*, 2018, **6**, 13628–13643.
- 21 Y.-T. Liao, V. C. Nguyen, N. Ishiguro, A. P. Young, C.-K. Tsung and K. C. W. Wu, *Appl. Catal., B*, 2020, **270**, 118805.
- 22 J. Wang, Y. Xu, B. Ding, Z. Chang, X. Zhang, Y. Yamauchi and K. C.-W. Wu, *Angew. Chem., Int. Ed.*, 2018, **57**, 2894–2898.
- 23 Y. Liu, X. Qin, S. Zhang, G. Liang, F. Kang, G. Chen and B. Li, *ACS Appl. Mater. Interfaces*, 2018, **10**, 26264–26273.
- 24 Y. Zhang and Q. Wei, *J. Electroanal. Chem.*, 2016, **781**, 401–409.
- 25 N. L. W. Septiani, A. G. Saputro, Y. V. Kaneti, A. L. Maulana, F. Fathurrahman, H. Lim, B. Yulianto, Nugraha, H. K. Dipojono, D. Golberg and Y. Yamauchi, *ACS Appl. Nano Mater.*, 2020, **3**, 8982–8996.
- 26 X. Yang, V. Salles, Y. V. Kaneti, M. Liu, M. Maillard, C. Journet, X. Jiang and A. Brioude, *Sens. Actuators, B*, 2015, **220**, 1112–1119.
- 27 X. Yang, W. Wang, C. Wang, H. Xie, H. Fu, X. An, X. Jiang and A. Yu, *Powder Technol.*, 2018, **339**, 408–418.
- 28 Y. V. Kaneti, N. L. Wulan Septiani, I. Saptiama, X. Jiang, B. Yulianto, M. J. A. Shiddiky, N. Fukumitsu, Y.-M. Kang, D. Golberg and Y. Yamauchi, *J. Mater. Chem. A*, 2019, **7**, 3415–3425.
- 29 A. Mirzaei, G.-J. Sun, J. K. Lee, C. Lee, S. Choi and H. W. Kim, *Ceram. Int.*, 2017, **43**, 5247–5254.
- 30 Z. Li, H. Li, Z. Wu, M. Wang, J. Luo, H. Torun, P. Hu, C. Yang, M. Grundmann, X. Liu and Y. Fu, *Mater. Horiz.*, 2019, **6**, 470–506.
- 31 M. Al-Hashem, S. Akbar and P. Morris, *Sens. Actuators, B*, 2019, **301**, 126845.
- 32 H. Yuan, S. A. A. Aljneibi, J. Yuan, Y. Wang, H. Liu, J. Fang, C. Tang, X. Yan, H. Cai, Y. Gu, S. J. Pennycook, J. Tao and D. Zhao, *Adv. Mater.*, 2019, **31**, 1807161.
- 33 Z. Li, Z. Yao, A. A. Haidry, T. Plecenik, L. Xie, L. Sun and Q. Fatima, *Int. J. Hydrogen Energy*, 2018, **43**, 21114–21132.
- 34 D. Ponnusamy, A. K. Prasad and S. J. M. A. Madanagurusamy, *Microchim. Acta*, 2016, **183**, 311–317.
- 35 Q. Ren, Y.-Q. Cao, D. Arulraj, C. Liu, D. Wu, W.-M. Li and A.-D. Li, *J. Electrochem. Soc.*, 2020, **167**, 067528.
- 36 M. Zhao, M. H. Wong, H. C. Man and C. W. Ong, *Sens. Actuators, B*, 2017, **249**, 624–631.
- 37 A. Sanger, A. Kumar, A. Kumar and R. Chandra, *Sens. Actuators, B*, 2016, **234**, 8–14.
- 38 K. Sankarasubramanian, P. Soundarajan, T. Logu, K. Sethuraman and K. Ramamurthi, *New J. Chem.*, 2018, **42**, 1457–1466.
- 39 A. Shanmugasundaram, P. Basak, L. Satyanarayana and S. V. Manorama, *Sens. Actuators, B*, 2013, **185**, 265–273.
- 40 M. B. Rahmani, M. H. Yaacob and Y. M. Sabri, *Sens. Actuators, B*, 2017, **251**, 57–64.
- 41 M. Kumar, V. Bhatt, A. C. Abhyankar, J.-H. Yun and H.-J. Jeong, *Int. J. Hydrogen Energy*, 2020, **45**, 15011–15025.
- 42 A. Simo, B. Mwakikunga, B. T. Sone, B. Julies, R. Madjoe and M. Maaza, *Int. J. Hydrogen Energy*, 2014, **39**, 8147–8157.
- 43 P. Arifin, M. A. Mustajab, S. Haryono, D. R. Adhika and A. A. Nugraha, *Mater. Res. Express*, 2019, **6**, 076313.
- 44 S. Haviar, J. Čapek, Š. Batková, N. Kumar, F. Dvořák, T. Duchoň, M. Fialová and P. Zeman, *Int. J. Hydrogen Energy*, 2018, **43**, 22756–22764.
- 45 O. Lupan, V. Postica, F. Labat, I. Ciofini, T. Pauporté and R. Adelung, *Sens. Actuators, B*, 2018, **254**, 1259–1270.



- 46 O. Alev, E. Şennik and Z. Z. Öztürk, *J. Alloys Compd.*, 2018, **749**, 221–228.
- 47 C.-H. Wu, Z. Zhu, H.-M. Chang, Z.-X. Jiang, C.-Y. Hsieh and R.-J. Wu, *J. Alloys Compd.*, 2020, **814**, 151815.
- 48 L. Huo, X. Yang, Z. Liu, X. Tian, T. Qi, X. Wang, K. Yu, J. Sun and M. Fan, *Sens. Actuators, B*, 2017, **244**, 694–700.
- 49 Y. Wang, Z. Zhao, Y. Sun, P. Li, J. Ji, Y. Chen, W. Zhang and J. Hu, *Sens. Actuators, B*, 2017, **240**, 664–673.
- 50 S. M. Majhi, P. Rai and Y.-T. Yu, *ACS Appl. Mater. Interfaces*, 2015, **7**, 9462–9468.
- 51 A. A. Haidry, L. Xie, Z. Wang and Z. Li, *Appl. Surf. Sci.*, 2020, **500**, 144219.
- 52 K. Thummavichai, Y. Xia and Y. Zhu, *Appl. Sci.*, 2017, **88**, 281–324.
- 53 Y. Luo, C. Zhang, B. Zheng, X. Geng and M. Debliquy, *Int. J. Hydrogen Energy*, 2017, **42**, 20386–20397.
- 54 A. Mirzaei, H. R. Yousefi, F. Falsafi, M. Bonyani, J.-H. Lee, J.-H. Kim, H. W. Kim and S. S. Kim, *Int. J. Hydrogen Energy*, 2019, **44**, 20552–20571.
- 55 M. M. Y. A. Alsaif, S. Balendhran, M. R. Field, K. Latham, W. Wlodarski, J. Z. Ou and K. Kalantar-zadeh, *Sens. Actuators, B*, 2014, **192**, 196–204.
- 56 A. Nandi, P. Nag, D. Panda, S. Dhar, S. M. Hossain, H. Saha and S. Majumdar, *ACS Omega*, 2019, **4**, 11053–11065.
- 57 X. Xia, W. Wu, Z. Wang, Y. Bao, Z. Huang and Y. Gao, *Sens. Actuators, B*, 2016, **234**, 192–200.
- 58 V. S. Nguyen, V. Jubera, A. Garcia and H. Debéda, *Appl. Surf. Sci.*, 2015, **357**, 14–21.
- 59 U. T. Nakate, G. H. Lee, R. Ahmad, P. Patil, Y.-B. Hahn, Y. T. Yu and E.-k. Suh, *Int. J. Hydrogen Energy*, 2018, **43**, 22705–22714.
- 60 U. T. Nakate, R. Ahmad, P. Patil, Y. T. Yu and Y.-B. Hahn, *Appl. Surf. Sci.*, 2020, **506**, 144971.
- 61 C. Yan, B. Huy Le and D. J. Kang, *J. Mater. Chem. A*, 2014, **2**, 5394–5398.
- 62 Y. Wang, B. Liu, D. Cai, H. Li, Y. Liu, D. Wang, L. Wang, Q. Li and T. Wang, *Sens. Actuators, B*, 2014, **201**, 351–359.
- 63 P. Li, Z. Xiong, S. Zhu, M. Wang, Y. Hu, H. Gu, Y. Wang and W. Chen, *Int. J. Hydrogen Energy*, 2017, **42**, 30186–30192.
- 64 O. Lupan, V. Cretu, V. Postica, M. Ahmadi, B. R. Cuenya, L. Chow, I. Tiginyanu, B. Viana, T. Pauporté and R. Adelung, *Sens. Actuators, B*, 2016, **223**, 893–903.
- 65 G. Korotcenkov, V. Brinzari, S. H. Han and B. K. Cho, *Mater. Chem. Phys.*, 2016, **175**, 188–199.
- 66 G. Kumar, X. Li, Y. Du, Y. Geng and X. Hong, *J. Alloys Compd.*, 2019, **798**, 467–477.
- 67 L. Chen, X. He, Y. Liang, Y. Sun, Z. Zhao and J. Hu, *J. Mater. Sci.: Mater. Electron.*, 2016, **27**, 11331–11338.
- 68 X. Wei, X. Yang, T. Wu, S. Wu, W. Li, X. Wang and Z. Chen, *Int. J. Hydrogen Energy*, 2017, **42**, 24580–24586.
- 69 A. Sanger, A. Kumar, A. Kumar, J. Jaiswal and R. Chandra, *Sens. Actuators, B*, 2016, **236**, 16–26.
- 70 R. Zhou, X. Lin, D. Xue, F. Zong, J. Zhang, X. Duan, Q. Li and T. Wang, *Sens. Actuators, B*, 2018, **260**, 900–907.
- 71 Z. Wang, S. Huang, G. Men, D. Han and F. Gu, *Sens. Actuators, B*, 2018, **262**, 577–587.
- 72 X.-T. Yin, W.-D. Zhou, J. Li, Q. Wang, F.-Y. Wu, D. Dastan, D. Wang, H. Garmestani, X.-M. Wang and Ş. Ṫalu, *J. Alloys Compd.*, 2019, **805**, 229–236.
- 73 I. Fratoddi, A. Macagnano, C. Battocchio, E. Zampetti, I. Venditti, M. V. Russo and A. Bearzotti, *Nanoscale*, 2014, **6**, 9177–9184.
- 74 M. T. Hosseinejad, M. Ghoranneviss, M. R. Hantehzadeh and E. Darabi, *J. Alloys Compd.*, 2016, **689**, 740–750.
- 75 O. Lupan, L. Chow, T. Pauporté, L. K. Ono, B. Roldan Cuenya and G. Chai, *Sens. Actuators, B*, 2012, **173**, 772–780.
- 76 Z. Zhang, C. Yin, L. Yang, J. Jiang and Y. Guo, *J. Alloys Compd.*, 2019, **785**, 819–825.
- 77 D. Sett and D. Basak, *Sens. Actuators, B*, 2017, **243**, 475–483.
- 78 A. Renitta and K. Vijayalakshmi, *Sens. Actuators, B*, 2016, **237**, 912–923.
- 79 G. Singh, N. Kohli and R. C. Singh, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 2257–2266.
- 80 C. Zhao, B. Huang, E. Xie, J. Zhou and Z. Zhang, *Sens. Actuators, B*, 2015, **207**, 313–320.
- 81 H. Liu, D. Ding, C. Ning and Z. Li, *Nanotechnology*, 2011, **23**, 015502.
- 82 K. Vijayalakshmi and A. Renitta, *Ceram. Int.*, 2015, **41**, 14315–14325.
- 83 B. Mondal, B. Basumatari, J. Das, C. Roychaudhury, H. Saha and N. Mukherjee, *Sens. Actuators, B*, 2014, **194**, 389–396.
- 84 J.-H. Lee, J.-Y. Kim, J.-H. Kim, A. Mirzaei, H. W. Kim and S. S. Kim, *Int. J. Hydrogen Energy*, 2019, **44**, 27499–27510.
- 85 J. Hu, Y. Sun, Y. Xue, M. Zhang, P. Li, K. Lian, S. Zhuyikov, W. Zhang and Y. Chen, *Sens. Actuators, B*, 2018, **257**, 124–135.
- 86 H. Kheel, G.-J. Sun, J. K. Lee, A. Mirzaei, S. Choi and C. Lee, *Met. Mater. Int.*, 2017, **23**, 214–219.
- 87 S. Park, G.-J. Sun, H. Kheel, S. Choi and C. Lee, *Mater. Res. Bull.*, 2016, **82**, 136–141.
- 88 D. Jung, M. Han and G. S. Lee, *Sens. Actuators, B*, 2014, **204**, 596–601.
- 89 S.-J. Liu, Y. Yuan, S.-L. Zheng, J.-H. Zhang and Y. Wang, *Dalton Trans.*, 2015, **44**, 11360–11367.
- 90 M. Mansha, A. Qurashi, N. Ullah, F. O. Bakare, I. Khan and Z. H. Yamani, *Ceram. Int.*, 2016, **42**, 11490–11495.
- 91 H. Ren, C. Gu, S. W. Joo, J. Zhao, Y. Sun and J. Huang, *Sens. Actuators, B*, 2018, **266**, 506–513.
- 92 V. Singh Bhati, A. Nathani, A. Nigam, C. S. Sharma and M. Kumar, *Sens. Actuators, B*, 2019, **299**, 126980.
- 93 S. M. Kim, H. J. Kim, H. J. Jung, J.-Y. Park, T. J. Seok, Y.-H. Choa, T. J. Park and S. W. Lee, *Adv. Funct. Mater.*, 2019, **29**, 1807760.
- 94 D. Zhang, Y. e. Sun, C. Jiang and Y. Zhang, *Sens. Actuators, B*, 2017, **242**, 15–24.
- 95 K. Hassan and G.-S. Chung, *Sens. Actuators, B*, 2017, **239**, 824–833.
- 96 V. S. Bhati, S. Ranwa, S. Rajamani, K. Kumari, R. Raliya, P. Biswas and M. Kumar, *ACS Appl. Mater. Interfaces*, 2018, **10**, 11116–11124.
- 97 S. Dhall, K. Sood and R. Nathawat, *Int. J. Hydrogen Energy*, 2017, **42**, 8392–8398.



- 98 T. Yang, Y. Liu, H. Wang, Y. Duo, B. Zhang, Y. Ge, H. Zhang and W. Chen, *J. Mater. Chem. C*, 2020, **8**, 7272–7299.
- 99 Z. Meng, R. M. Stolz, L. Mendecki and K. A. Mirica, *Chem. Rev.*, 2019, **119**, 478–598.
- 100 H. Nazemi, A. Joseph, J. Park and A. Emadi, *Sensors*, 2019, **19**, 1285.
- 101 A. Mirzaei, S. S. Kim and H. W. Kim, *J. Hazard. Mater.*, 2018, **357**, 314–331.
- 102 K. Inyawilert, A. Wisitsoraat, A. Tuantranont, S. Phanichphant and C. Liewhiran, *Sens. Actuators, B*, 2017, **240**, 1141–1152.
- 103 S. Zeb, X. Peng, G. Yuan, X. Zhao, C. Qin, G. Sun, Y. Nie, Y. Cui and X. Jiang, *Sens. Actuators, B*, 2020, **305**, 127435.
- 104 X. Liu, T. Ma, N. Pinna and J. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1702168.
- 105 M. Weber, J.-Y. Kim, J.-H. Lee, J.-H. Kim, I. Iatsunskyi, E. Coy, P. Miele, M. Bechelany and S. S. Kim, *J. Mater. Chem. A*, 2019, **7**, 8107–8116.
- 106 H. Xu, W. Li, R. Han, T. Zhai, H. Yu, Z. Chen, X. Wu, J. Wang and B. Cao, *Sens. Actuators, B*, 2018, **262**, 70–78.
- 107 O. Alev, E. Şennik, N. Kılınc and Z. Z. Öztürk, *Procedia Eng.*, 2015, **120**, 1162–1165.
- 108 P. Li, D. Zhang, C. Jiang, X. Zong and Y. Cao, *Biosens. Bioelectron.*, 2017, **98**, 68–75.
- 109 S. Pandey, *J. Sci.: Adv. Mater. Devices*, 2016, **1**, 431–453.
- 110 A. A. Baharuddin, B. C. Ang, A. S. M. A. Haseeb, Y. C. Wong and Y. H. Wong, *Mater. Sci. Semicond. Process.*, 2019, **103**, 104616.
- 111 N. Sihar, T. Y. Tiong, C. F. Dee, P. C. Ooi, A. A. Hamzah, M. A. Mohamed and B. Y. Majlis, *Nanoscale Res. Lett.*, 2018, **13**, 150.
- 112 M. Sinha, R. Mahapatra, B. Mondal, T. Maruyama and R. Ghosh, *J. Phys. Chem. C*, 2016, **120**, 3019–3025.
- 113 J. Huang and Q. Wan, *Sensors*, 2009, **9**, 9903–9924.
- 114 Z. Li, S. Yan, Z. Wu, H. Li, J. Wang, W. Shen, Z. Wang and Y. Fu, *Int. J. Hydrogen Energy*, 2018, **43**, 22746–22755.
- 115 Z. Wu, Z. Li, H. Li, M. Sun, S. Han, C. Cai, W. Shen and Y. Fu, *ACS Appl. Mater. Interfaces*, 2019, **11**, 12761–12769.
- 116 X.-T. Yin, J. Li, D. Dastan, W.-D. Zhou, H. Garmestani and F. M. Alamgir, *Sens. Actuators, B*, 2020, **319**, 128330.
- 117 F. Rasch, V. Postica, F. Schütt, Y. K. Mishra, A. S. Nia, M. R. Lohe, X. Feng, R. Adelung and O. Lupan, *Sens. Actuators, B*, 2020, **320**, 128363.
- 118 U. T. Nakate, R. Ahmad, P. Patil, Y. Wang, K. S. Bhat, T. Mahmoudi, Y. T. Yu, E.-k. Suh and Y.-B. Hahn, *J. Alloys Compd.*, 2019, **797**, 456–464.
- 119 X. Liu, S. Cheng, H. Liu, S. Hu, D. Zhang and H. Ning, *Sensors*, 2012, **12**, 9635–9665.
- 120 C. Zhang, Y. Luo, J. Xu and M. Debliquy, *Sens. Actuators, A*, 2019, **289**, 118–133.
- 121 N. Y. Chan, M. Zhao, J. Huang, K. Au, M. H. Wong, H. M. Yao, W. Lu, Y. Chen, C. W. Ong, H. L. W. Chan and J. Dai, *Adv. Mater.*, 2014, **26**, 5962–5968.
- 122 H.-P. Looock and P. D. Wentzell, *Sens. Actuators, B*, 2012, **173**, 157–163.
- 123 D. Degler, U. Weimar and N. Barsan, *ACS Sens.*, 2019, **4**, 2228–2249.
- 124 H. Ji, W. Zeng and Y. Li, *Nanoscale*, 2019, **11**, 22664–22684.
- 125 E. Lee, Y. S. Yoon and D.-J. Kim, *ACS Sens.*, 2018, **3**, 2045–2060.
- 126 H.-J. Kim and J.-H. Lee, *Sens. Actuators, B*, 2014, **192**, 607–627.
- 127 J. Yi, H. Zhang, Z. Zhang and D. Chen, *Sens. Actuators, B*, 2018, **268**, 456–464.
- 128 C. O. Park and S. A. Akbar, *J. Mater. Sci.*, 2003, **38**, 4611–4637.
- 129 N. Barsan and U. Weimar, *J. Electroceram.*, 2001, **7**, 143–167.
- 130 S. Park, *Curr. Appl. Phys.*, 2016, **16**, 1263–1269.
- 131 A. Katoch, S.-W. Choi, H. W. Kim and S. S. Kim, *J. Hazard. Mater.*, 2015, **286**, 229–235.
- 132 S. Yang, G. Lei, Z. Lan, W. Xie, B. Yang, H. Xu, Z. Wang and H. Gu, *Int. J. Hydrogen Energy*, 2019, **44**, 7725–7733.
- 133 H. Fu, X. Yang, X. An, W. Fan, X. Jiang and A. Yu, *Sens. Actuators, B*, 2017, **252**, 103–115.
- 134 Y. V. Kaneti, X. Zhang, M. Liu, D. Yu, Y. Yuan, L. Aldous and X. Jiang, *Sens. Actuators, B*, 2016, **230**, 581–591.
- 135 A. Umar, H. Y. Ammar, R. Kumar, T. Almas, A. A. Ibrahim, M. S. AlAssiri, M. Abaker and S. Baskoutas, *Int. J. Hydrogen Energy*, 2020, **45**, 26388–26401.
- 136 Y. Zhang, W. Zeng and Y. Li, *Mater. Res. Bull.*, 2018, **107**, 139–146.
- 137 S. Zeb, G. Sun, Y. Nie, Y. Cui and X. Jiang, *Sens. Actuators, B*, 2020, **321**, 128439.
- 138 T. Wagner, S. Haffer, C. Weinberger, D. Klaus and M. Tiemann, *Chem. Soc. Rev.*, 2013, **42**, 4036–4053.
- 139 M.-H. Sun, S.-Z. Huang, L.-H. Chen, Y. Li, X.-Y. Yang, Z.-Y. Yuan and B.-L. Su, *Chem. Soc. Rev.*, 2016, **45**, 3479–3563.
- 140 U. T. Nakate, R. Ahmad, P. Patil, K. S. Bhat, Y. Wang, T. Mahmoudi, Y. T. Yu, E.-k. Suh and Y.-B. Hahn, *Int. J. Hydrogen Energy*, 2019, **44**, 15677–15688.
- 141 Y. Zhang, W. Zeng and Y. Li, *Ceram. Int.*, 2019, **45**, 6043–6050.
- 142 J. Zhang, Z. Qin, D. Zeng and C. Xie, *Phys. Chem. Chem. Phys.*, 2017, **19**, 6313–6329.
- 143 Y. Shen, W. Wang, A. Fan, D. Wei, W. Liu, C. Han, Y. Shen, D. Meng and X. San, *Int. J. Hydrogen Energy*, 2015, **40**, 15773–15779.
- 144 N. Lavanya, C. Sekar, E. Fazio, F. Neri, S. G. Leonardi and G. Neri, *Int. J. Hydrogen Energy*, 2017, **42**, 10645–10655.
- 145 S. Yang, Z. Wang, Y. Hu, Y. Cai, R. Huang, X. Li, Z. Huang, Z. Lan, W. Chen and H. Gu, *Sens. Actuators, B*, 2018, **260**, 21–32.
- 146 Y. Zou, J. He, Y. Hu, R. Huang, Z. Wang and Q. Gu, *RSC Adv.*, 2018, **8**, 16897–16901.
- 147 S. Yang, Z. Wang, Y. Hu, X. Luo, J. Lei, D. Zhou, L. Fei, Y. Wang and H. Gu, *ACS Appl. Mater. Interfaces*, 2015, **7**, 9247–9253.
- 148 P. V. Shinde, B. G. Ghule, S. F. Shaikh, N. M. Shinde, S. S. Sangale, V. V. Jadhav, S.-Y. Yoon, K. H. Kim and R. S. Mane, *J. Alloys Compd.*, 2019, **802**, 244–251.
- 149 D. Abubakar, N. M. Ahmed, S. Mahmud and N. A. Algadri, *Mater. Res. Express*, 2017, **4**, 075009.



- 150 A. M. Soleimanpour, S. V. Khare and A. H. Jayatissa, *ACS Appl. Mater. Interfaces*, 2012, **4**, 4651–4657.
- 151 Y. Dong, D. Tang and C. Li, *Sci. China: Technol. Sci.*, 2014, **57**, 2153–2160.
- 152 I. H. Kadhim and H. Abu Hassan, *J. Electron. Mater.*, 2017, **46**, 1419–1426.
- 153 I. H. Kadhim, H. A. Hassan and Q. N. Abdullah, *Nano-Micro Lett.*, 2016, **8**, 20–28.
- 154 A. Hazra, S. Das, J. Kanungo, C. K. Sarkar and S. Basu, *Sens. Actuators, B*, 2013, **183**, 87–95.
- 155 N. Pradeep, C. Venkatachalaiah, U. Venkatraman, C. Santhosh, A. Bhatnagar, S. K. Jeong and A. N. Grace, *Microchim. Acta*, 2017, **184**, 3349–3355.
- 156 T. Stoycheva, F. E. Annanouch, I. Gràcia, E. Llobet, C. Blackman, X. Correig and S. Vallejos, *Sens. Actuators, B*, 2014, **198**, 210–218.
- 157 Y. Shen, X. Cao, B. Zhang, D. Wei, J. Ma, W. Liu, C. Han and Y. Shen, *J. Alloys Compd.*, 2014, **593**, 271–274.
- 158 X. San, G. Wang, B. Liang, Y. Song, S. Gao, J. Zhang and F. Meng, *J. Alloys Compd.*, 2015, **622**, 73–78.
- 159 O. K. Alexeeva and V. N. Fateev, *Int. J. Hydrogen Energy*, 2016, **41**, 3373–3386.
- 160 E. Turgut, Ö. Çoban, S. Sarıtaş, S. Tüzemen, M. Yıldırım and E. Gür, *Appl. Surf. Sci.*, 2018, **435**, 880–885.
- 161 Z. Cai, B. Liu, X. Zou and H.-M. Cheng, *Chem. Rev.*, 2018, **118**, 6091–6133.
- 162 J. Yu, J. Li, W. Zhang and H. Chang, *Chem. Sci.*, 2015, **6**, 6705–6716.
- 163 M. Parashar, V. K. Shukla and R. Singh, *J. Mater. Sci.: Mater. Electron.*, 2020, **31**, 3729–3749.
- 164 A. V. Vinogradov and V. V. Vinogradov, *RSC Adv.*, 2014, **4**, 45903–45919.
- 165 S. Mersagh Dezfuli and M. Sabzi, *Appl. Phys. A: Mater. Sci. Process.*, 2019, **125**, 557.
- 166 N. Xue, Q. Zhang, S. Zhang, P. Zong and F. Yang, *Sensors*, 2017, **17**, 2351.
- 167 H. J. Sharma, N. D. Sonwane and S. B. Kondawar, *Fibers Polym.*, 2015, **16**, 1527–1532.
- 168 L. A. Mercante, R. S. Andre, L. H. C. Mattoso and D. S. Correa, *ACS Appl. Nano Mater.*, 2019, **2**, 4026–4042.
- 169 J. Xue, T. Wu, Y. Dai and Y. Xia, *Chem. Rev.*, 2019, **119**, 5298–5415.
- 170 Y. Wang, I. Ramos and J. J. Santiago-Aviles, *IEEE Sens. J.*, 2007, **7**, 1347–1348.
- 171 X. Lu, C. Wang and Y. J. S. Wei, *Small*, 2009, **5**, 2349–2370.
- 172 R. Ab Kadir, Z. Li, A. Z. Sadek, R. Abdul Rani, A. S. Zoolfakar, M. R. Field, J. Z. Ou, A. F. Chrimes and K. Kalantar-zadeh, *J. Phys. Chem. C*, 2014, **118**, 3129–3139.
- 173 C.-L. Zhang and S.-H. Yu, *Chem. Soc. Rev.*, 2014, **43**, 4423–4448.
- 174 W. Shi, S. Song and H. Zhang, *Chem. Soc. Rev.*, 2013, **42**, 5714–5743.
- 175 A. V. Nikam, B. L. V. Prasad and A. A. Kulkarni, *CrystEngComm*, 2018, **20**, 5091–5107.
- 176 X. Wu, S. Xiong, Z. Mao, Y. Gong, W. Li, B. Liu, S. Hu and X. Long, *J. Alloys Compd.*, 2017, **704**, 117–123.
- 177 A. Mirzaei and G. Neri, *Sens. Actuators, B*, 2016, **237**, 749–775.
- 178 A. Shanmugasundaram, B. Ramireddy, P. Basak, S. V. Manorama and S. Srinath, *J. Phys. Chem. C*, 2014, **118**, 6909–6921.
- 179 P. G. Choi, N. Izu, N. Shirahata and Y. Masuda, *ACS Omega*, 2018, **3**, 14592–14596.
- 180 X. Zhou, Z. Wang, X. Xia, G. Shao, K. Homewood and Y. Gao, *ACS Appl. Mater. Interfaces*, 2018, **10**, 28199–28209.
- 181 H. M. M. Munasinghe Arachchige, D. Zappa, N. Poli, N. Gunawardhana, N. H. Attanayake and E. Comini, *Nano-materials*, 2020, **10**, 935.
- 182 Y.-T. Wang, W.-T. Whang and C.-H. Chen, *ACS Appl. Mater. Interfaces*, 2015, **7**, 8480–8487.
- 183 Y. K. Mishra, G. Modi, V. Cretu, V. Postica, O. Lupan, T. Reimer, I. Paulowicz, V. Hrkac, W. Benecke, L. Kienle and R. Adelung, *ACS Appl. Mater. Interfaces*, 2015, **7**, 14303–14316.
- 184 Q. A. Drmash, Z. H. Yamani and M. K. Hossain, *Sens. Actuators, B*, 2017, **248**, 868–877.
- 185 M. Kumar, V. Bhatt, A. Kumar and J.-H. Yun, *Mater. Lett.*, 2019, **240**, 13–16.
- 186 M. Kumar, V. Bhatt, J. Kim, A. C. Abhyankar, H.-J. Chung, K. Singh, Y. B. Cho, Y. J. Yun, K. S. Lim and J.-H. Yun, *Sens. Actuators, B*, 2021, **326**, 128839.
- 187 J.-H. Lee, J.-Y. Kim, J.-H. Kim and S. S. Kim, *Sensors*, 2019, **19**, 726.
- 188 M. Thepnurat, T. Chairuang Sri, N. Hongsith, P. Ruankham and S. Choo Pun, *ACS Appl. Mater. Interfaces*, 2015, **7**, 24177–24184.
- 189 F. H. Tian, C. Gong, Y. Peng and X. Xue, *Sens. Actuators, B*, 2017, **244**, 655–663.
- 190 Y. Chen, X. Wang, C. Shi, L. Li, H. Qin and J. Hu, *Sens. Actuators, B*, 2015, **220**, 279–287.
- 191 S. Sun, X. Zhang, J. Cui, Q. Yang and S. Liang, *Nanoscale*, 2019, **11**, 15739–15762.
- 192 H. Zhang, T. Tao, X. Li, Y. Bao, X. Xia, M. Lourenço, K. Homewood, Z. Huang and Y. Gao, *Int. J. Hydrogen Energy*, 2020, **45**, 18057–18065.
- 193 X. Zhou, H. Zhang, Z. Wang, X. Xia, Y. Bao, K. Homewood, G. Shao, Z. Huang and Y. Gao, *Int. J. Hydrogen Energy*, 2019, **44**, 20606–20615.
- 194 M. Hübner, C. E. Simion, A. Tomescu-Stănoiu, S. Pokhrel, N. Bârsan and U. Weimar, *Sens. Actuators, B*, 2011, **153**, 347–353.
- 195 O. Lupan, V. Postica, N. Ababii, M. Hoppe, V. Cretu, I. Tiginyanu, V. Sontea, T. Pauporté, B. Viana and R. Adelung, *Microelectron. Eng.*, 2016, **164**, 63–70.
- 196 I. Karaduman, T. Çorlu, M. A. Yıldırım, A. Ateş and S. Acar, *J. Electron. Mater.*, 2017, **46**, 4017–4023.
- 197 N. D. Hoa, P. Van Tong, C. M. Hung, N. Van Duy and N. Van Hieu, *Int. J. Hydrogen Energy*, 2018, **43**, 9446–9453.
- 198 O. Lupan, V. Postica, N. Wolff, J. Su, F. Labat, I. Ciofini, H. Cavers, R. Adelung, O. Polonskyi, F. Faupel, L. Kienle, B. Viana and T. Pauporté, *ACS Appl. Mater. Interfaces*, 2019, **11**, 32115–32126.



- 199 C. Liewhiran, N. Tamaekong, A. Tuantranont, A. Wisitsoraat and S. Phanichphant, *Mater. Chem. Phys.*, 2014, **147**, 661–672.
- 200 D. Kaewsiri, K. Inyawilert, A. Wisitsoraat, A. Tuantranont, S. Phanichphant and C. Liewhiran, *Sens. Actuators, B*, 2020, **316**, 128132.
- 201 K. Vijayalakshmi, A. Renitta and A. Monamary, *J. Mater. Sci.: Mater. Electron.*, 2018, **29**, 21023–21032.
- 202 K. Vijayalakshmi, A. Renitta and K. Karthick, *Ceram. Int.*, 2014, **40**, 6171–6177.
- 203 Y. Li, D. Deng, N. Chen, X. Xing, X. Liu, X. Xiao and Y. Wang, *J. Alloys Compd.*, 2017, **710**, 216–224.
- 204 X. Yang, W. Wang, J. Xiong, L. Chen and Y. Ma, *Int. J. Hydrogen Energy*, 2015, **40**, 12604–12609.
- 205 Y. Bao, P. Wei, X. Xia, Z. Huang, K. Homewood and Y. Gao, *Sens. Actuators, B*, 2019, **301**, 127143.
- 206 Y. Hong, Z. Lin, Z. Chen and G. Wang, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 8837–8843.
- 207 E. Fazio, S. G. Leonardi, M. Santoro, N. Donato, G. Neri and F. Neri, *Sens. Actuators, B*, 2018, **262**, 79–85.
- 208 S. Zhang, C. Yin, L. Yang, Z. Zhang and Z. Han, *Sens. Actuators, B*, 2019, **283**, 399–406.
- 209 E. S. Babu and S. K. Hong, *Superlattices Microstruct.*, 2015, **82**, 349–356.
- 210 Z. Li, Q. Yang, Y. Wu, Y. He, J. Chen and J. Wang, *Int. J. Hydrogen Energy*, 2019, **44**, 8659–8668.
- 211 J.-H. Kim, A. Mirzaei, H. Woo Kim and S. S. Kim, *Sens. Actuators, B*, 2019, **284**, 628–637.
- 212 Y. Hou and A. H. Jayatissa, *Sens. Actuators, B*, 2014, **204**, 310–318.
- 213 C. Li, M. Iqbal, J. Lin, X. Luo, B. Jiang, V. Malgras, K. C. W. Wu, J. Kim and Y. Yamauchi, *Acc. Chem. Res.*, 2018, **51**, 1764–1773.
- 214 D. Kang, T. W. Kim, S. R. Kubota, A. C. Cardiel, H. G. Cha and K.-S. Choi, *Chem. Rev.*, 2015, **115**, 12839–12887.
- 215 J. Leng, Z. Wang, J. Wang, H.-H. Wu, G. Yan, X. Li, H. Guo, Y. Liu, Q. Zhang and Z. Guo, *Chem. Soc. Rev.*, 2019, **48**, 3015–3072.
- 216 J.-S. Park, J. K. Kim, J. H. Hong, J. S. Cho, S.-K. Park and Y. C. Kang, *Nanoscale*, 2019, **11**, 19012–19057.
- 217 F. Meierhofer, H. Li, M. Gockeln, R. Kun, T. Grieb, A. Rosenauer, U. Fritsching, J. Kiefer, J. Birkenstock, L. Mädler and S. Pokhrel, *ACS Appl. Mater. Interfaces*, 2017, **9**, 37760–37777.
- 218 T. Samerjai, N. Tamaekong, K. Wetchakun, V. Kruefu, C. Liewhiran, C. Siri Wong, A. Wisitsoraat and S. Phanichphant, *Sens. Actuators, B*, 2012, **171–172**, 43–61.
- 219 Y.-L. Huang, H.-J. Liu, C.-H. Ma, P. Yu, Y.-H. Chu and J.-C. Yang, *Chin. J. Phys.*, 2019, **60**, 481–501.
- 220 M. Horprathum, T. Srichaiyaperk, B. Samransuksamer, A. Wisitsoraat, P. Eiamchai, S. Limwichean, C. Chananonawathorn, K. Aiempnanakit, N. Nuntawong, V. Patthanasettakul, C. Oros, S. Porntheeraphat, P. Songsiririthigul, H. Nakajima, A. Tuantranont and P. Chindaudom, *ACS Appl. Mater. Interfaces*, 2014, **6**, 22051–22060.
- 221 R. Jolly Bose, N. Illyaskutty, K. S. Tan, R. S. Rawat, M. V. Matham, H. Kohler and V. P. Mahadevan Pillai, *EPL*, 2016, **114**, 66002.
- 222 Y. Wang, B. Liu, S. Xiao, H. Li, L. Wang, D. Cai, D. Wang, Y. Liu, Q. Li and T. Wang, *J. Mater. Chem. A*, 2015, **3**, 1317–1324.
- 223 B. Liu, D. Cai, Y. Liu, D. Wang, L. Wang, Y. Wang, H. Li, Q. Li and T. Wang, *Sens. Actuators, B*, 2014, **193**, 28–34.
- 224 S. Xiao, B. Liu, R. Zhou, Z. Liu, Q. Li and T. Wang, *Sens. Actuators, B*, 2018, **254**, 966–972.
- 225 E. Şennik, O. Alev and Z. Z. Öztürk, *Sens. Actuators, B*, 2016, **229**, 692–700.
- 226 M. Jiao, N. Van Duy, N. V. Chien, N. D. Hoa, N. Van Hieu, K. Hjort and H. Nguyen, *Int. J. Hydrogen Energy*, 2017, **42**, 16294–16304.
- 227 R. K. Chava, S.-Y. Oh and Y.-T. Yu, *CrystEngComm*, 2016, **18**, 3655–3666.
- 228 Q. A. Drmash and Z. H. Yamani, *Appl. Surf. Sci.*, 2016, **375**, 57–64.
- 229 K. Inyawilert, A. Wisitsoraat, C. Liewhiran, A. Tuantranont and S. Phanichphant, *Appl. Surf. Sci.*, 2019, **475**, 191–203.
- 230 N. Van Toan, N. Viet Chien, N. Van Duy, H. Si Hong, H. Nguyen, N. Duc Hoa and N. Van Hieu, *J. Hazard. Mater.*, 2016, **301**, 433–442.
- 231 F. Wang, K. Hu, H. Liu, Q. Zhao, K. Wang and Y. Zhang, *Int. J. Hydrogen Energy*, 2020, **45**, 7234–7242.
- 232 P. Nag, S. Majumdar, A. Bumajdad and P. S. Devi, *RSC Adv.*, 2014, **4**, 18512–18521.
- 233 J.-A. Woo, D.-T. Phan, Y. W. Jung and K.-J. Jeon, *Int. J. Hydrogen Energy*, 2017, **42**, 18754–18761.
- 234 M. Zhao, J. X. Huang and C. W. Ong, *Sens. Actuators, B*, 2014, **191**, 711–718.
- 235 S.-J. Choi, S. Chattopadhyay, J. J. Kim, S.-J. Kim, H. L. Tuller, G. C. Rutledge and I.-D. Kim, *Nanoscale*, 2016, **8**, 9159–9166.
- 236 F. E. Annanouch, Z. Haddi, M. Ling, F. Di Maggio, S. Vallejos, T. Vilic, Y. Zhu, T. Shujah, P. Umek, C. Bittencourt, C. Blackman and E. Llobet, *ACS Appl. Mater. Interfaces*, 2016, **8**, 10413–10421.
- 237 H. Kim, Y. Pak, Y. Jeong, W. Kim, J. Kim and G. Y. Jung, *Sens. Actuators, B*, 2018, **262**, 460–468.
- 238 J.-H. Kim, A. Mirzaei, H. W. Kim and S. S. Kim, *Sens. Actuators, B*, 2019, **297**, 126693.
- 239 Z. Tang, Y. Zhang, X. Deng, Y. Dai, W. Zhang, F. Fan, B. Qing, C. Zhu, J. Fan and Y. Shi, *Dalton Trans.*, 2018, **47**, 15331–15337.
- 240 T.-R. Rashid, D.-T. Phan and G.-S. Chung, *Sens. Actuators, B*, 2014, **193**, 869–876.
- 241 M. F. B. Alam, D.-T. Phan and G.-S. Chung, *Mater. Lett.*, 2015, **156**, 113–117.
- 242 M. Shahabuddin, A. Umar, M. Tomar and V. Gupta, *Int. J. Hydrogen Energy*, 2017, **42**, 4597–4609.
- 243 N. X. Thai, N. Van Duy, N. Van Toan, C. M. Hung, N. Van Hieu and N. D. Hoa, *Int. J. Hydrogen Energy*, 2020, **45**, 2418–2428.
- 244 W. P. Chen, Y. Xiong, Y. S. Li, P. Cui, S. S. Guo, W. Chen, Z. L. Tang, Z. Yan and Z. Zhang, *Int. J. Hydrogen Energy*, 2016, **41**, 3307–3312.



- 245 A. A. Haidry, L. Xie, Z. Wang, A. Zavabeti, Z. Li, T. Plecenik, M. Gregor, T. Roch and A. Plecenik, *ACS Sens.*, 2019, **4**, 2997–3006.
- 246 X. Yang, H. Fu, L. Zhang, X. An, S. Xiong, X. Jiang and A. Yu, *Sens. Actuators, B*, 2019, **286**, 483–492.
- 247 C. Lupan, R. Khaledialidusti, A. K. Mishra, V. Postica, M.-I. Terasa, N. Magariu, T. Pauporté, B. Viana, J. Drewes, A. Vahl, F. Faupel and R. Adelung, *ACS Appl. Mater. Interfaces*, 2020, **12**, 24951–24964.
- 248 G. Singh, Virpal and R. C. Singh, *Sens. Actuators, B*, 2019, **282**, 373–383.
- 249 S. Pati, P. Banerji and S. B. Majumder, *RSC Adv.*, 2015, **5**, 61230–61238.
- 250 K. Vijayalakshmi and K. Karthick, *Int. J. Hydrogen Energy*, 2014, **39**, 7165–7172.
- 251 K. Karthick, D. Srinivasan and J. B. Christopher, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 11979–11986.
- 252 A. Monamary and K. Vijayalakshmi, *J. Mater. Sci.: Mater. Electron.*, 2018, **29**, 5316–5326.
- 253 V. S. Bhati, S. Ranwa, M. Fanetti, M. Valant and M. Kumar, *Sens. Actuators, B*, 2018, **255**, 588–597.
- 254 J. Luo, Y. Li, X. Mo, Y. Xu and Q. Zeng, *RSC Adv.*, 2017, **7**, 29844–29853.
- 255 A. Katoch, J.-H. Kim, Y. J. Kwon, H. W. Kim and S. S. Kim, *ACS Appl. Mater. Interfaces*, 2015, **7**, 11351–11358.
- 256 A. Katoch, Z. U. Abideen, H. W. Kim and S. S. Kim, *ACS Appl. Mater. Interfaces*, 2016, **8**, 2486–2494.
- 257 J.-H. Lee, J.-Y. Kim, A. Mirzaei, H. W. Kim and S. S. Kim, *Nanomaterials*, 2018, **8**, 902.
- 258 H. Liu, F. Wang, K. Hu, B. Zhang, L. He and Q. Zhou, *Nanomaterials*, 2019, **9**, 1250.
- 259 D. E. Motaung, G. H. Mhlongo, P. R. Makgwane, B. P. Dhonge, F. R. Cummings, H. C. Swart and S. S. Ray, *Sens. Actuators, B*, 2018, **254**, 984–995.
- 260 T. L. Ruwer, J. Venturini, S. Khan and C. P. Bergmann, *Mater. Lett.*, 2020, **265**, 127429.
- 261 H. Fan, S. Xu, X. Cao, D. Liu, Y. Yin, H. Hao, D. Wei and Y. Shen, *Appl. Surf. Sci.*, 2017, **400**, 440–445.
- 262 K. Mondal and A. Sharma, *RSC Adv.*, 2016, **6**, 94595–94616.
- 263 S. Park, G.-J. Sun, H. Kheel, S. K. Hyun, C. Jin and C. Lee, *Met. Mater. Int.*, 2016, **22**, 156–162.
- 264 S. Park, *Mater. Lett.*, 2019, **234**, 315–318.
- 265 S. Park, S. Park, S. Lee, H. W. Kim and C. Lee, *Sens. Actuators, B*, 2014, **202**, 840–845.
- 266 H. Xun, Z. Zhang, A. Yu and J. Yi, *Sens. Actuators, B*, 2018, **273**, 983–990.
- 267 S. Park, H. Kheel, G.-J. Sun, H. W. Kim, T. Ko and C. Lee, *Met. Mater. Int.*, 2016, **22**, 730–736.
- 268 S. Pati, P. Banerji and S. B. Majumder, *Int. J. Hydrogen Energy*, 2014, **39**, 15134–15141.
- 269 M. H. Raza, N. Kaur, E. Comini and N. Pinna, *ACS Appl. Mater. Interfaces*, 2020, **12**, 4594–4606.
- 270 Z. Zhang, C. Yin, L. Yang, W. Jia, J. Zhou, H. Xu and D. Cao, *Ceram. Int.*, 2017, **43**, 6693–6699.
- 271 X. Wu, S. Xiong, Z. Mao, S. Hu and X. J. C. E. J. Long, *Chem. – Eur. J.*, 2017, **23**, 7969–7975.
- 272 S.-W. Choi, A. Katoch, J.-H. Kim and S. S. Kim, *ACS Appl. Mater. Interfaces*, 2014, **6**, 17723–17729.
- 273 W.-D. Zhou, D. Dastan, J. Li, X.-T. Yin and Q. Wang, *Nanomaterials*, 2020, **10**, 785.
- 274 S. Mao, G. Lu and J. Chen, *J. Mater. Chem. A*, 2014, **2**, 5573–5579.
- 275 A. Lee, J. Park, K. S. Choi, J. Lee, I. Yoo, I. S. Cho, B. Ahn, H. Seo, J.-Y. Choi and H. K. Yu, *Carbon*, 2017, **125**, 221–226.
- 276 T. Kamal, *J. Alloys Compd.*, 2017, **729**, 1058–1063.
- 277 M. Zhang, Y. Zhen, F. Sun and C. Xu, *Mater. Sci. Eng., B*, 2016, **209**, 37–44.
- 278 D. Dutta, S. K. Hazra, J. Das, C. K. Sarkar and S. Basu, *Sens. Actuators, B*, 2015, **212**, 84–92.
- 279 M. Bhatnagar, S. Dhall, V. Kaushik, A. Kaushal and B. R. Mehta, *Sens. Actuators, B*, 2017, **246**, 336–343.
- 280 D. Kathiravan, B.-R. Huang and A. Saravanan, *ACS Appl. Mater. Interfaces*, 2017, **9**, 12064–12072.
- 281 K. Anand, O. Singh, M. P. Singh, J. Kaur and R. C. Singh, *Sens. Actuators, B*, 2014, **195**, 409–415.
- 282 D. Zhang, N. Yin, C. Jiang and B. Xia, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 2763–2768.
- 283 Z. U. Abideen, H. W. Kim and S. S. Kim, *Chem. Commun.*, 2015, **51**, 15418–15421.
- 284 S. Majumdar, P. Nag and P. S. Devi, *Mater. Chem. Phys.*, 2014, **147**, 79–85.
- 285 W. C. Huang, H. J. Tsai, T. C. Lin, W. C. Weng, Y. C. Chang, J. L. Chiu, J.-J. Lin, C. F. Lin, Y.-S. Lin and H. Chen, *Ceram. Int.*, 2018, **44**, 12308–12314.
- 286 E. Singh, M. Meyyappan and H. S. Nalwa, *ACS Appl. Mater. Interfaces*, 2017, **9**, 34544–34586.
- 287 X. Feng, W. Chen and L. Yan, *RSC Adv.*, 2016, **6**, 80106–80113.
- 288 S. W. Lee, W. Lee, Y. Hong, G. Lee and D. S. Yoon, *Sens. Actuators, B*, 2018, **255**, 1788–1804.
- 289 J. Wang, S. Rathi, B. Singh, I. Lee, H.-I. Joh and G.-H. Kim, *ACS Appl. Mater. Interfaces*, 2015, **7**, 13768–13775.
- 290 A. Esfandiari, A. Irajizad, O. Akhavan, S. Ghasemi and M. R. Gholami, *Int. J. Hydrogen Energy*, 2014, **39**, 8169–8179.
- 291 T. D. Nguyen, C. D. Nguyen, N. V. Hieu and M. J. J. A. M. I. MacLachlan, *Adv. Mater.*, 2018, **5**, 1800269.
- 292 Y. Luo and C. Zhang, *J. Alloys Compd.*, 2018, **747**, 550–557.
- 293 A. Kumar, A. Sanger, A. Kumar and R. Chandra, *RSC Adv.*, 2017, **7**, 39666–39675.
- 294 A. Monamary and K. Vijayalakshmi, *Ceram. Int.*, 2018, **44**, 22957–22962.
- 295 J.-H. Lee, J.-H. Kim, J.-Y. Kim, A. Mirzaei, H. W. Kim and S. S. Kim, *Sensors*, 2019, **19**, 4276.
- 296 S. Nasirian and H. Milani Moghaddam, *Appl. Surf. Sci.*, 2015, **328**, 395–404.
- 297 K. Hassan, A. S. M. I. Uddin, F. Ullah, Y. S. Kim and G.-S. Chung, *Mater. Lett.*, 2016, **176**, 232–236.
- 298 F. Fan, J. Zhang, J. Li, N. Zhang, R. Hong, X. Deng, P. Tang and D. Li, *Sens. Actuators, B*, 2017, **241**, 895–903.
- 299 S. Dhall and N. Jaggi, *Sens. Actuators, B*, 2015, **210**, 742–747.



- 300 Q. A. Drmosh, A. H. Hendi, M. K. Hossain, Z. H. Yamani, R. A. Moqbel, A. Hezam and M. A. Gondal, *Sens. Actuators, B*, 2019, **290**, 666–675.
- 301 V. Galstyan, A. Ponzoni, I. Kholmanov, M. M. Natile, E. Comini, S. Nematov and G. Sberveglieri, *ACS Sens.*, 2019, **4**, 2094–2100.
- 302 H. Abdollahi, M. Samkan and M. M. Hashemi, *New J. Chem.*, 2019, **43**, 19253–19264.
- 303 L. S. K. Achary, B. Maji, A. Kumar, S. P. Ghosh, J. P. Kar and P. Dash, *Int. J. Hydrogen Energy*, 2020, **45**, 5073–5085.
- 304 A. Saravanan, B.-R. Huang, J. P. Chu, A. Prasannan and H.-C. Tsai, *Sens. Actuators, B*, 2019, **292**, 70–79.
- 305 Y. Kumaresan, H. Kim, Y. Jeong, Y. Pak, S. Cho, R. Lee, N. Lim and G. Y. Jung, *IEEE Electron Device Lett.*, 2017, **38**, 1735–1738.
- 306 Z. Li, Z. Yao, A. A. Haidry, T. Plecenik, B. Grancic, T. Roch, M. Gregor and A. Plecenik, *J. Alloys Compd.*, 2019, **806**, 1052–1059.
- 307 A. Harley-Trochimczyk, T. Pham, J. Chang, E. Chen, M. A. Worsley, A. Zettl, W. Mickelson and R. Maboudian, *Adv. Funct. Mater.*, 2016, **26**, 433–439.
- 308 M. Das and S. Roy, *Mater. Sci. Semicond. Process.*, 2021, **121**, 105332.
- 309 J. Huang, J. Zhu, W. Sun and J. Ji, *ACS Appl. Mater. Interfaces*, 2020, **12**, 47048–47058.
- 310 S. Nasirian and S. Y. Razavi, *Mater. Sci. Eng., B*, 2017, **224**, 40–47.
- 311 S. R. Jamnani, H. M. Moghaddam, S. G. Leonardi and G. Neri, *Synth. Met.*, 2020, **268**, 116493.
- 312 S. Nasirian and H. Milani Moghaddam, *Int. J. Hydrogen Energy*, 2014, **39**, 630–642.
- 313 T.-R. Rashid, D.-T. Phan and G.-S. Chung, *Sens. Actuators, B*, 2013, **185**, 777–784.
- 314 I. Darmadi, F. A. A. Nugroho and C. Langhammer, *ACS Sens.*, 2020, **5**, 3306–3327.

