


 Cite this: *RSC Adv.*, 2021, **11**, 27152

# AgNPs decorated 3D bionic silicon nanoglass arrays pattern with high-density hot-spots for SERS sensing *via* green galvanic displacement without additives†

 Li Wang,<sup>‡a</sup> Jian Huang,<sup>‡\*a</sup> Mei-Juan Su,<sup>a</sup> Jin-Di Wu<sup>a</sup> and Weisheng Liu<sup>ID</sup><sup>\*b</sup>

Surface-enhanced Raman scattering (SERS) sensing has always been considered as a kind of high-efficiency analysis technique in different areas. Herein, we report a AgNPs decorated 3D bionic silicon (Si) nanoglass SERS substrate with higher sensitivity and specificity by green galvanic displacement. The Si nanoglass arrays are directly grown on a Si substrate *via* catalyst-assisted vapor–liquid–solid (VLS) growth and subsequent plasma interaction. AgNPs were rapidly immobilized on Si nanoglass arrays without any organic reagents, and avoiding the interference signal of additives. The AgNPs decorated 3D bionic silicon nanoglass arrays not only possess a larger specific surface area (loading more reporter molecules), but also provide a potential distribution and arrangement for plentiful hot spots. Using Rhodamine 6G (R6G) as a probe molecule, the prepared SERS substrates exhibited great potential for high-sensitivity SERS sensing, and pushed the limit of detection (LOD) down to 0.1 pM. A higher Raman analytical enhancement factor (AEF,  $3.3 \times 10^7$ ) was obtained, which was two magnitudes higher than our previous Ag micro–nano structures. Additionally, the practicality and reliability of our 3D bionic SERS substrates were confirmed by quantitative analysis of the spiked Sudan I in environmental water, with a wide linear range (from  $10^{-10}$  M to  $10^{-6}$  M) and low detection limit (0.1 nM).

 Received 23rd June 2021  
 Accepted 29th July 2021

DOI: 10.1039/d1ra04874k

[rsc.li/rsc-advances](http://rsc.li/rsc-advances)

## 1. Introduction

As one of the most significant analytical tools, SERS has attracted intensive attention because it can provide meaningful finger-print information for identifying analytes in complex samples.<sup>1–5</sup> It has been used in various fields including food safety, environmental monitoring, bioanalysis, and homeland security, owing to its high sensitivity and specificity.<sup>6–8</sup> As we all know, a remarkable SERS effect is critically dependent on the reliable SERS-active substrates such as different noble metal nanoparticles, nanowires, nanofilms, and nanorods.<sup>9</sup> However, these traditional one-component noble metal substrates are often expensive, additives are overused, and they lack high-density hot-spots, restricting their practical applications.<sup>10</sup>

Recently, semiconductor nano-materials combined with noble metal have drawn remarkable interest as promising SERS-

active substrates and meet extensive practical requirements (inexpensive, less organic additives, and abundant hot spots) very well.<sup>11–13</sup> They perfectly combine the superior properties of noble metal and semiconductor nano-materials.

But it should also be emphasized that, even a small amount of organic additives (such as reductants, surfactants, coupling agents, and other functional agents), originating from the synthesis of noble metal/semiconductor (Au–ZnO,<sup>14,15</sup> Ag–Fe<sub>3</sub>O<sub>4</sub>,<sup>16–19</sup> Pt–TiO<sub>2</sub>,<sup>20,21</sup> *et al.*), still diminish the plasmonic activity of the SERS substrates and causing a loss of probe binding sites. Besides, they may bring extra interferential bands, if the target analytes have similar Raman spectral response with the organic additives.

Therefore, it is crucial to devise a green strategy for the preparation of noble metal/semiconductor SERS-active substrates without organic additives. If the noble metal nanoparticles could be directly *in situ* reduced to their support semiconductor, the clean surface without organic additives would exhibit a significantly improved background, and the SERS activity would also be tremendously enhanced.

Silicon (Si) nanomaterials have been extensively studied as semiconductor SERS substrates for ultra-sensitive detection due to their attractive properties.<sup>22</sup> Among them, silicon nanospheres,<sup>23</sup> nanowires,<sup>24</sup> and nanoarrays<sup>25</sup> are successfully exploited as the commonly used silicon scaffolding in an

<sup>a</sup>College of Chemistry and Chemical Engineering, Xi'an Shiyu University, Xi'an, Shaanxi 710065, China. E-mail: [jhuang@xsyu.edu.cn](mailto:jhuang@xsyu.edu.cn)

<sup>b</sup>Key Laboratory of Nonferrous Metals Chemistry and Resources Utilization of Gansu Province, College of Chemistry and Chemical Engineering, Lanzhou University, Lanzhou 730000, P. R. China. E-mail: [liuws@lzu.edu.cn](mailto:liuws@lzu.edu.cn)

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d1ra04874k

‡ These two authors have made an equal contribution to this work and hold joint first authorship.



assortment of applications. First, silicon scaffolding is an easy-to-prepare and relatively cheap semiconductor substrate. Next, silicon scaffolding is neutral for most analytes, and not cause a luminescence background. Further, silicon scaffolding is not merely a solid substrate to deposit AgNPs, it can play the role of reducing agent.

Previous studies<sup>26</sup> have suggested that the reductive surfaces possess weak reducibility, and can react with the oxidative noble-metal salt in aqueous solution with no need of any organic additives. By this *in situ* redox reaction, it should be possible to directly grow noble metal particles on silicon scaffolding surface.

The first reports on fabrication of AgNPs on silicon scaffolding for SERS sensing were published more than a dozen years ago.<sup>27</sup> In these studies the AgNPs were synthesized on the silicon wafers by immersing in a mixed aqueous solution of silver nitrate and hydrofluoric acid. Although this method realized the *in situ* reduction in a simple way, the composite SERS substrate only generates an acceptable hot spots density.

Notably, the three-dimensional (3D) silicon nanostructures with abundant hierarchical scaffoldings are far superior than the reported one- and two-dimensional silicon scaffolding materials, owing to their potential for the hot spots arrangement along the third dimension. As a good candidate for silicon nanostructures SERS substrates, 3D bionic silicon nanoglass arrays, which are composed of silicon nanoneedles with controllable length, diameter, apex angle, and potential reducing power, were highly desirable. The 3D bionic silicon nanoglass arrays have been chosen as the silicon scaffolding for AgNPs immobilizing for the following reasons. The unique optical reflectance and light absorption properties of semiconducting nanowire arrays are promising building blocks for photonics.<sup>28,29</sup>

The pre-processed 3D bionic silicon nanoglass arrays can provide the necessary reduction potential for the green decoration of AgNPs, by virtue of their hydrogen-terminal surfaces.

In this process, the 3D bionic silicon nanoglass arrays serves as both the source of electrons that reduce the Ag ion in solution and the scaffolds for the loading of AgNPs. The average size and distribution of the decorated AgNPs can be effectively controlled by metal salt concentration and deposition time. Furthermore, catalyst AuNPs residues in the top of silicon nanoglass can also be used for electromagnetic enhancement (similar to lightning rod effect).

On the basis of the above considerations, here, we report a simple green route for synthesizing AgNPs decorated 3D bionic silicon nanoglass arrays, pattern with high-density hot-spots without additives. The 3D bionic silicon nanoglass arrays were prepared by PECVD method using the shrinkage and consumption of AuNPs catalyst under plasma conditioning. They not only have a large specific surface area (can adsorb more probe molecules), but also have the potential to load abundant AgNPs. The subsequent *in situ* redox employed the hydrogen fluoride etched silicon surfaces perfectly, and realized the reduction of Ag<sup>+</sup> (*viz.* green galvanic displacement). No foreign reducing agents and other organic additives were used, avoiding the introduction of impurities. All of the above selected reagents and exquisitely prepared steps ensured the substrates surfaces were free of organic contaminants,

exhibiting an improved background signal. For instance, based on the clean surfaces and high-density hot spots, the AgNPs decorated 3D bionic silicon nanoglass arrays had achieved a highly sensitive detection of Rhodamine 6G (R6G) with a detection limit of 0.1 pM. The analytical enhancement factor (AEF) was up to  $3.3 \times 10^7$ , which was much higher than pure silver or silicon nanoarrays. Meanwhile, the spiked Sudan I in environmental water, with a low concentration of 0.1 nM can easily be determined, much lower than the maximum residue limits (MRLs). These results demonstrated the great potential of the as-prepared AgNPs/silicon nanoglass arrays SERS substrates in the fields of food safety, environmental monitoring, bioanalysis, and so on.

## 2. Results and discussion

### 2.1 Preparation of silicon nanoglass arrays and its growth mechanism

Fig. 1 exhibits the SEM and TEM images of the 3D bionic silicon nanoglass arrays at different magnifications. As shown in Fig. 1a and b, the as-prepared 3D bionic silicon nanoglass arrays are high-density needle-like, and uniform over a large area. The cross section image (Fig. 1c) reveals that the bionic silicon nanoglass arrays are partial vertical to the original silicon substrates. Through the preliminary statistical analysis, we found that the diameters of single silicon nanoneedle are in the range from 30 nm to 130 nm. In Fig. 1d, the TEM image indicates that as-grown 3D bionic silicon nanoglass is needle-like and has a crystalline silicon core, which is coated by a thin layer silicon dioxide. Normally, the thickness of oxide layer is no more than 5 nm and can be removed by immersing into the hydrofluoric acid aqueous solution (with a concentration of about 5% by volume) for 2 min at room temperature.

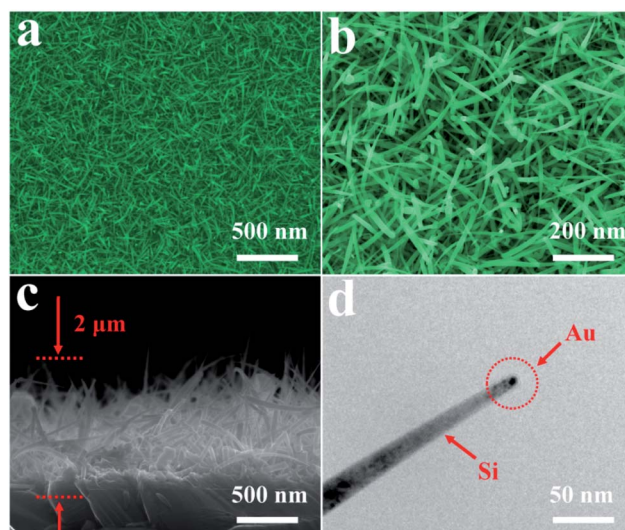


Fig. 1 SEM and TEM images of 3D bionic silicon nanoglass arrays grown on silicon substrates using PECVD method at different magnifications: (a) low-magnification SEM micrographs, (b) high-magnification SEM micrographs, (c) cross section SEM view, and (d) high-magnification TEM micrographs.



The as-processed silicon nanograss is stored in vacuumed glass tubes to prevent any contact with air/oxygen and used for the subsequent decoration of AgNPs (by *in situ* chemical reduction *via* galvanic displacement). Furthermore, in Fig. 1d, further research found that the AuNPs catalysts are distributed mainly at the top of silicon nanograss and their external surface. Obviously, this is a typical vapor-liquid-solid (VLS) growth processes, which are universal pathway for the preparation of semiconductor nanomaterial *via* noble metal catalysts. The related growth mechanisms are investigated and shown in Fig. 2.

As shown in Fig. 2, high-quality 3D bionic silicon nanograss is grown on the commonly used p-type Si(100) substrates by the PECVD method, under the VLS mechanism. It is generally known that the morphology, shape, length, and diameter of the as-prepared nanograss is dependent on the type and size of catalysts during their growth. For example, Fig. 2a represents the catalyst layer of gold film deposited on the surface of silicon substrate. Usually, the gold layer is prepared by magnetron sputtering or ion sputtering, and the optimum thickness of the gold layer is found to be 5–10 nm. Then, a mild annealing experiment had been carried out under vacuum (0.1–10 mbar) in the 400–600 °C temperature range. After annealing under vacuum, which led to gold catalyst layer changes in both the morphology and component. Owing to the dewetting effect, the gold layer gradually cracked and turned into catalyst islands. It is a major step in the growth process of silicon nanograss arrays, which transformed the metal catalyst thin films into nanosize islands (Fig. 2b). During this process, the single-component catalyst nanoislands became a double-component catalyst alloy droplets (*via* gold-silicon alloys catalysts).

Many investigations indicate that the silicon comes from thermolysis of Si(100) substrates.<sup>30</sup> Besides, small solid gold nanoparticles also catalyse the decomposition of the vaporous silane precursor, forming a liquid silicon-metal alloy. The corresponding morphology characters and component analysis of catalyst nanoislands were also investigated in our previous research. In the VLS mechanism, as more of silane precursor is added to the system, crystalline silicon nanograss arrays are excreted from the alloy due to oversaturation (Fig. 2c). Based on this mechanism, the diameter and length of the silicon nanograss arrays are directly correlated to the size of catalytic alloy droplets.

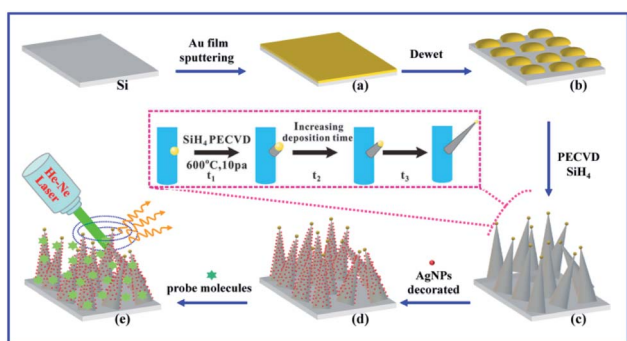


Fig. 2 Schematic drawing of AgNPs decorated 3D bionic silicon nanograss arrays and its growth mechanism: (a) deposition of gold film, (b) dewetting evolution, (c) initial growth, (d) decoration of AgNPs, and (e) probe adsorption, SERS sensing detection.

In our experimental conditions, the droplets on the tip of the silicon nanograss arrays are gradually exhausted due to thermal evaporation, plasma etching, and linear loss into the sidewall spreading, then the needle-like silicon nanograss gradually grow (Fig. 2c – inset, Fig. S2 in ESI†). The gold nanoparticles from the catalyst droplets wet the nanowires sidewalls, eventually consuming the droplets. The changeable size and shrinking droplets would directly affect the morphology of as-growth silicon nanograss arrays. That is to say, the shrinkage of silicon-gold catalyst droplets leads to gradual tapering nanograss arrays taking place (Fig. 2d).

## 2.2 Silver nanoparticles decorated silicon nanograss arrays and its characterization

The previous researches indicate that silver nanoparticles exhibited greater advantages in comparison to gold and copper nanoparticles owing to their stronger plasmonic ability, and can be more applicable to large-scale production.<sup>31</sup> We herein report a facile method for the fabrication of the novel SERS substrates based on 3D bionic silicon nanograss arrays decorated with silver nanoparticles by a simple green *in situ* galvanic displacement and their application for the detection of ultra-low concentration of different analytes (Fig. 2d and e).

As we all know that the as-prepared silicon nanograss arrays usually have an oxide outer-layer due to the oxidation process of during or after their growth. However, when the silicon nanograss arrays are immersed into hydrofluoric acid solution (5%, mass fraction), the outer oxide layer can be easily removed, and Si-H bonds ( $\text{SiH}_x$ ) forms on their outer surfaces. The hydrogen-terminated silicon nanograss arrays (H-SINGAs) is a kind of moderate *in situ* reducing reagent, which can fast direct reduce silver ions ( $\text{Ag}^+$ ) to elemental silver (Ag) at room temperature in aqueous solution. Then the HF-treated silicon nanograss ( $\text{SiH}_x$ ) is reoxidized to  $\text{SiO}_x$  surface and dissolves into the water, which exists in the form of  $\text{Si}(\text{OH})_4$ . In this process, the outer surface  $\text{SiH}_x$  ( $x = 1-3$ ) species are consumed quickly, which may facilitate the total redox reaction. Many studies show that hydrogen-terminated silicon nanostructures possess mild reduction properties and can be considered as the main reducing source for the decoration of noble metal nanoparticles.<sup>32</sup> Although a plausible reaction mechanism can be deduced from the corresponding reaction products, the actual reaction between hydrogen-terminated silicon nanostructures and silver ions in aqueous solution is still a complex surface electrochemistry process.

Detailed information about the surface electronic structure of the obtained AgNPs/silicon nanograss arrays are investigated by X-ray photoelectron spectroscopy (XPS) with an X-ray Al-K $\alpha$  source in Fig. 3a. Fig. 3b records the evolution of the Si 2p electronic state. In the Si 2p region of the as-prepared AgNPs/silicon nanograss arrays, there is an intense peak (99.38 eV) at  $\sim 4.2$  eV above the binding energy of the Si 2p (a broad peak centered at 103.68 eV). It has been established that surface Si can be readily oxidized by electronegative elements through several oxidation states, and the binding energy increase associated with a unit increase in oxidation state (Si-O bond) is  $\sim 0.68$  eV. In Fig. 3b, the intense peak is thus attributed to  $\text{SiO}_2$  on the outer layer of the 3D bionic silicon nanograss arrays. In



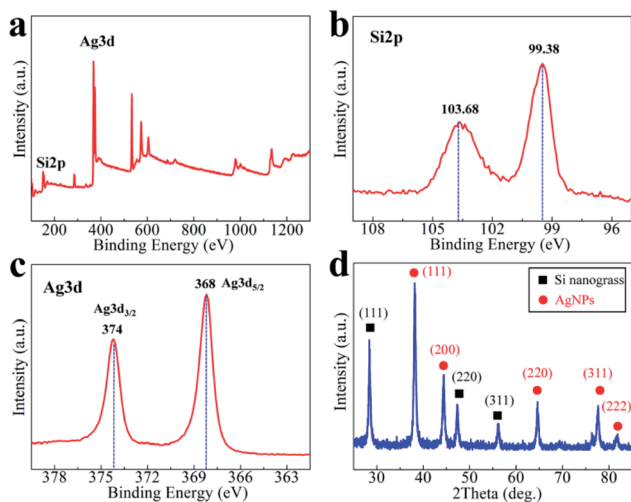


Fig. 3 XPS spectra and XRD pattern of the as-prepared AgNPs decorated 3D bionic silicon nanoglass arrays: (a) XPS survey spectrum, (b) Si 2p, (c) Ag 3d, and (d) XRD analysis.

In addition, there is a broad peak centered at 103.68 eV, which is the O2s peak of oxygen in SiO<sub>2</sub>. Fig. 3c shows the XPS spectra of Ag 3d of the AgNPs. The binding energies of Ag 3d<sub>5/2</sub> and Ag 3d<sub>3/2</sub> are detected at 368 and 374 eV, respectively, which are well consistent with those of Ag metal, confirming that they are Ag nanoparticles on the surface of silicon nanoglass arrays. The XRD pattern (Fig. 3d) demonstrates that the products have a high degree of crystallization. The main peaks of AgNPs are located at  $2\theta$  of 38.3°, 44.6°, 64.3° and 77.7°, corresponding to (111), (200), (220), and (311) planes, respectively. The results of XRD have also exhibited the nature of Ag(0), which strongly indicates that the AgNPs have been successfully modified.

In addition, the HAADF STEM image with corresponding elemental maps results are shown in Fig. 4 (10<sup>-4</sup> M AgNO<sub>3</sub> aqueous solution treated silicon nanoglass arrays), which also shows that they are Ag nanoparticles on the surface of 3D bionic silicon nanoglass arrays. Previous research indicates that the surface morphology as well as the average size of the noble metal nanoparticles is the most significant factors in constructing ultrasensitive SERS substrates.<sup>33</sup> By choosing and controlling the available reaction parameters, we can improve the enhancement efficiency and obtain a reliable ultrasensitive SERS substrates. Generally, it is achieved by rationally tuning the experimental parameters including the reaction time, reaction temperature, and precursor concentration.

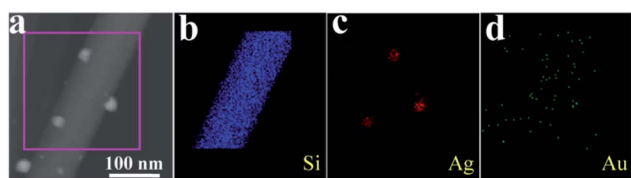


Fig. 4 (a) HAADF STEM image and corresponding elemental mapping images for (b) Si, (c) Ag, and (d) Au are presented in panels respectively. The gold originated from catalyst residue.

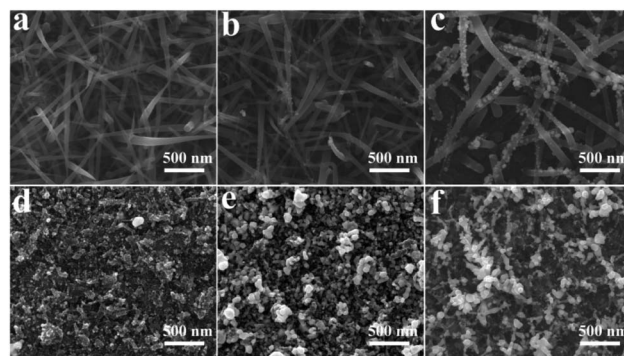


Fig. 5 SEM images of AgNPs decorated 3D bionic silicon nanoglass arrays in different reaction time: (a) 30 s, (b) 1 min, (c) 2 min, (d) 5 min, (e) 8 min, and (f) 10 min.

The effects of reaction time on the as-prepared substrate structures, morphologies, and SERS enhancement performances are investigated and shown in Fig. 5. The initial concentration of silver nitrate is up to 0.1 mM, the size evolution of silver nanoparticles is a time-dependent process. The morphology of AgNPs decorated silicon nanoglass differs in different reaction time.

It is obvious that the average diameter and density of AgNPs increases gradually through the different growing stages (in 2 min, Fig. 5a–c, by increasing the reaction time), and significant difference is found between these stages.

The overall result reveals that AgNPs begin to form in one minute, and nanoparticles begin to merge and coalesce with neighbouring nanowires to form composite SERS substrate in the first two minutes. However, with further extension of the reaction time, various continuous silver membrane and irregular silver nanoparticles form on the surface of the 3D bionic silicon nanoglass arrays owing to the excessive aggregation nanoparticles (Fig. 5d and e). Obviously, the small Ag seeds synthesized in the early stage which act as growth seeds of the Ag plates and Ag dendrites (Fig. 5f) by further prolonging the reaction time which will lead to a decreasing SERS intensity. Some big Ag plates (Fig. 5d), Ag particles (Fig. 5e) and Ag dendrites (Fig. 5f) are formed subsequently on the silicon substrate by the accumulation of time cycles, which also disturb the displacement of AgNPs. Based on the above studies, an optimal comprehensive reaction period (2 min) of silver decoration is summarized.

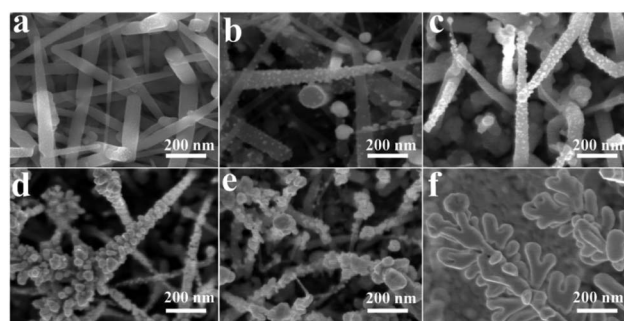
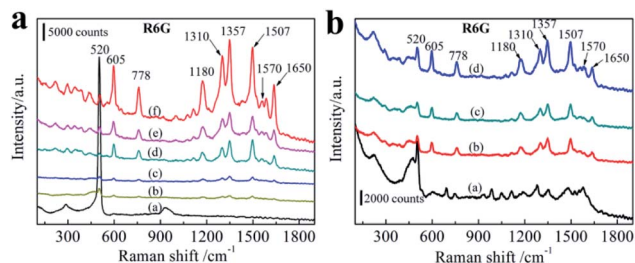


Fig. 6 SEM images of AgNPs decorated 3D bionic silicon nanoglass arrays in different initial AgNO<sub>3</sub> concentrations: (a) 10<sup>-6</sup> M, (b) 10<sup>-5</sup> M, (c) 10<sup>-4</sup> M, (d) 10<sup>-3</sup> M, (e) 10<sup>-2</sup> M, and (f) 10<sup>-1</sup> M.





**Fig. 7** (a) 3D bionic silicon nanoglass arrays treated in different  $\text{AgNO}_3$  concentration and their corresponding SERS spectra of R6G: curve (b)  $10^{-1}$  M, curve (c)  $10^{-5}$  M, curve (d)  $10^{-4}$  M, curve (e)  $10^{-2}$  M, curve (f)  $10^{-3}$  M, and curve (a) 0 M; (b) SERS spectra of R6G (incident power was 3.4 mW, data acquisition time was 10 s) at different concentration: curve (a)  $10^{-13}$  M, curve (b)  $10^{-11}$  M, curve (c)  $10^{-10}$  M, and curve (d)  $10^{-9}$  M.

Besides the reaction time, the decoration of AgNPs could also be tuned by different initial  $\text{AgNO}_3$  concentrations in the galvanic displacement reaction process. Fig. 6 shows the size evolution of AgNPs coated silicon nanoglass arrays by increasing the  $\text{AgNO}_3$  concentrations. It is observed that for all six silver ion concentrations, the average size of AgNPs is formed differently on the surface of silicon nanoglass. The size and number of AgNPs increases with increasing concentrations of  $\text{Ag}^+$  (from  $10^{-6}$  M to  $10^{-3}$  M, Fig. 6a–d), while a higher  $\text{Ag}^+$  concentration will cause irregular AgNPs (Fig. 6e) and Ag dendrites (Fig. 6f).

A rough statistical results display that the average sizes of AgNPs are about 5, 10, 15 and 40 nm for  $10^{-6}$  M,  $10^{-5}$  M,  $10^{-4}$  M and  $10^{-3}$  M concentration, respectively. Based on the different average size of AgNPs, which will cause much closer gap distance. Researches show that those 3D bionic silicon nanoglass arrays SERS substrates have different effectiveness in their SERS detection process.

### 2.3 SERS detection of R6G dye

In order to understand the relationship between SERS properties and average size of AgNPs, these six AgNPs decorated 3D bionic silicon nanoglass arrays SERS substrates treated in different initial  $\text{AgNO}_3$  concentrations (from  $10^{-1}$  M to  $10^{-6}$  M) are evaluated by detecting the Raman spectra of the same concentration R6G. By immersing the different AgNPs size substrates into same concentration R6G aqueous solution for 3 h, a sufficient coverage of monolayer is achieved on AgNPs surface. Then the excessive molecules adsorbed on substrates are rinsed in deionized water twice and dried in  $\text{N}_2$  before acquiring the SERS spectra. The AgNPs/needle-like silicon nanoglass arrays substrates prepared at different  $\text{AgNO}_3$  concentrations provide the similar R6G Raman spectra, except the intensity of main characteristic bands in Fig. 7a. The wave number from 550 to  $1700\text{ cm}^{-1}$ , the prominent Raman characteristic bands of R6G at 1180, 1310, 1357, 1507, 1570 and  $1650\text{ cm}^{-1}$  are all observed clearly in all of these substrates which agreed well with previously literature data. The peak at  $1180\text{ cm}^{-1}$  is associated with C–C in-plane bend modes, and the signals at 1310, 1357, 1507, 1570 and  $1650\text{ cm}^{-1}$  (between  $1300$  and  $1700\text{ cm}^{-1}$ ) are due to the symmetric in-plane C–C stretching vibrations of R6G.

Fig. 7a also shows that different initial  $\text{AgNO}_3$  concentration leads to different AgNPs size on the 3D bionic silicon nanoglass arrays, which also exhibited different SERS spectra. In the present case, the major source of hot spots originates from the morphology of the aggregate AgNPs. In Fig. 7a, with increasing of initial  $\text{AgNO}_3$  concentrations (curve c, d, and f), the AgNPs grow larger and aggregated, the SERS intensity also increases gradually. With further extension of initial  $\text{AgNO}_3$  concentrations (curve e, b), a continuous AgNPs membrane and Ag dendrites form on the surface of 3D bionic silicon nanoglass arrays due to the excessive aggregation of AgNPs on the surface, the SERS intensity begin to decrease evidently.

Curve f exhibits the strongest Raman spectra, whereas curve b and curve c provides the minimum SERS spectrum compared with the pure 3D bionic silicon nanoglass arrays (curve a). We can infer that a low density and a tiny size of AgNPs (SEM Fig. 6a) prepared at lower  $\text{AgNO}_3$  concentration ( $10^{-5}$  M) which lead to SERS intensity decrease evidently. The same question occurs in curve b, the decreasing SERS spectrum which due to the forming of Ag dendrites (SEM Fig. 6f) at higher  $\text{AgNO}_3$  concentration ( $10^{-1}$  M). Once the Ag dendrites grow together on the surface of 3D substrate, the Raman enhancement disappears obviously. Curve d and curve e shows secondary Raman spectra attribute to the particle coverage density of AgNPs. Experiments indicate that a proper  $\text{Ag}^+$  concentration creates a reasonable particle coverage density and size of AgNPs on the 3D bionic silicon nanoglass arrays.

The maximum SERS spectrum is shown on the curve f, SEM image shows that the substrate uniform AgNPs and close distance. A rough statistics shows the diameter is 40 nm and the gap distance is  $\sim 9.2$  nm.  $10^{-3}$  M  $\text{AgNO}_3$  concentration creates an optimal size and gap distance of AgNPs on the needle-like 3D bionic silicon nanoglass arrays. As the growth of diameter of AgNPs, the AgNPs begin to aggregate with narrow gaps between each other.

Fig. 7b shows the SERS spectra of R6G with a concentration gradients ranging from  $10^{-9}$  down to  $10^{-13}$  M. Prominent Raman spectra can be readily identified from concentration  $10^{-13}$  M and above. The Raman intensity of peak at  $1357\text{ cm}^{-1}$  decreases slowly with decreasing concentration of R6G, and is clearly observable as low as  $10^{-13}$  M. Our experiment results indicate that the 3D bionic silicon nanoglass arrays SERS substrate exhibits a lower detection limit (0.1 pM) and prominent SERS signal enhancement effect.

Furthermore, Raman enhancement factor (EF) can be estimated from the experiments (see ESI† the five main Raman bands at 614, 776, 1360, 1507 and  $1570\text{ cm}^{-1}$  are selected to calculate the average analytical enhancement factor (AEF)). On the basis of calculations, the AEF value of the AgNPs decorated 3D bionic silicon nanoglass arrays substrate is estimated to  $3.3 \times 10^7$  (several possible factors are located at ESI†).

### 2.4 Practical analysis of Sudan I in environmental water

Sudan I may be one of the most notorious industrial dyes that was implicated in hot chilli powder and the chilli products, because it have a toxic effect after oral intake on the body and



can lead to cause cancer in healthy adults.<sup>34–36</sup> Many countries are making rules to detect and fight Sudan I food pollution. Sensitive detection and quantification of Sudan I dye can play important roles in food supervision and inspection. In this regard, by using a highly sensitive and reproducible SERS substrates coupled with Raman spectrometer can provides a quick qualitative and quantitative analysis with easy sample preparation. In order to investigate whether our 3D bionic silicon nanograss arrays SERS substrate is highly sensitive for Sudan I detection or not, we prepare Sudan I aqueous solutions with different concentrations from  $10^{-5}$  to  $10^{-10}$  M and introduce them to the nanograss arrays for SERS detection.

Fig. 8a shows the representative SERS spectra of 3D bionic silicon nanograss arrays under different Sudan I concentrations and the Raman spectral intensities are decreased by diluting the Sudan I aqueous solutions. Obviously, the SERS spectra of all six Sudan I are consistent with their corresponding Raman spectra and contain all major characteristic peaks (see Table S1 in ESI†). A typical characteristic peak of Sudan I is clearly observed in each spectrum at  $1592\text{ cm}^{-1}$  which is assigned to the stretching vibration of the benzene ring.<sup>37,38</sup> These distinct characteristic peaks (at  $1592\text{ cm}^{-1}$ ) are observed even when the Sudan I concentration was down to  $10^{-10}$  M, demonstrating the high sensitivity of the as-prepared 3D silicon nanograss arrays. More importantly, it is much lower than the general default MRLs in many countries.<sup>39</sup> Such a low limit of detection enables the as-fabricated silicon nanograss arrays SERS substrates could be utilized toward the goal of trace analysis of illegal additive residues.<sup>40</sup> The increasing trend of SERS intensity with Sudan I concentration is further summarized in Fig. 8b for quantitative Sudan I sensing. And the intense SERS peak at  $1592\text{ cm}^{-1}$  in the spectra has been used as a calibration band. A gradual increase in SERS intensity is observed as the Sudan I concentration increasing. Above  $10^{-5}$  M, a step response is observed in signal intensity, indicating a stronger response to high concentration of Sudan I.

The calibration curve in the concentration ranging from  $10^{-10}$  to  $10^{-6}$  M is shown in Fig. 8b (inset). The linear equation is determined as  $\log I_{1592} = (0.2113 \pm 0.1577)\log C + (5.3580 \pm 0.1287)$ , ( $10^{-10} < C < 10^{-6}$ ) with the correlation coefficient  $R^2 = 0.992$ , where  $I_{1592}$  is the SERS intensity and  $C$  is the concentration of Sudan I dye. Based on these studies, the linear equation could provide an opening for fast ultra-sensitive

analysis of trace Sudan I in environmental water or illegal additive-contaminated products, which has the potential for a practical application. Moreover, the quantification of ultra-trace Sudan I residual in environment water is successfully carried out with the recovery of 89.4% to 112.0% (Table S2, ESI†). The stability and durability of the prepared substrates are investigated in Fig. S3 (ESI†), and exhibited a good stability in 15 days, a 19% decrement after one month. These results confirm the practical application of our 3D bionic SERS substrate with high accuracy and reliability.

## 3. Experimental

### 3.1 Chemical reagents

Silver nitrate ( $\text{AgNO}_3$ ) was purchased from Xi'an Dayu Chemicals Industry; hydrofluoric acid (HF) was purchased from Dongjiang Chemicals Company; silicon (Si) wafers were purchased from Luoyang Single Crystal Silicon Company; silane ( $\text{SiH}_4$ ) was purchased from Nanjing Specialty Gas Company; Rhodamine 6G (R6G), Sudan I was purchased from J&K Chemical in Beijing. All chemicals were used as received, without further purification. Milli-Q water ( $>18.0\text{ M}\Omega\text{ cm}$ ) was used throughout the experiments. The environmental water comes from the eastern lake of Xi'an.

### 3.2 Preparation of 3D bionic silicon nanograss arrays

The 3D bionic silicon nanograss arrays used in this work were prepared by plasma enhanced chemical vapor deposition (PECVD) method, because the PECVD technology can better meet the deposition requirements of high-efficiency, low-cost, large-scale, and homogeneous distribution. The growth of silicon nanograss was carried out in PECVD chamber based a gold-catalyzed vapor–liquid–solid (VLS) growth mechanism. The detailed experimental procedure for the preparation of silicon nanograss involved the following steps: first, the pre-cut silicon (100) chips ( $5\text{ cm} \times 5\text{ cm}$ ) were ultrasonically cleaned sequentially in alcohol, acetone, and deionized (DI) water each for 10 min, respectively, and then dried with nitrogen blow before they were placed into the vacuum chamber. Second, a thin gold layer of 10 nm was magnetron sputtered (or ion sputtering) on top of silicon substrates. Then, the substrates were loaded into the PECVD reactor, where alloy droplets were formed by vacuum annealing at a temperature of  $800\text{ }^\circ\text{C}$ . Finally, the growths of needle-like Si nanowires were performed for 1 h in a low pressure PECVD system by using silane as precursor. The reaction temperature can be selected from  $450$  to  $800\text{ }^\circ\text{C}$ , the base pressure of the vacuum chamber was  $4.0 \times 10^{-4}$  Pa, the silane (40 sccm, 10% in hydrogen) pressure was 250 mTorr when sputtering, and the radio frequency power was  $30\text{ mW cm}^{-2}$ .

### 3.3 AgNPs decorated 3D bionic silicon nanograss arrays and corresponding characterization

In order to obtain the H-terminal silicon nanograss arrays, the as-prepared 3D bionic silicon nanograss arrays substrates were immersed into a 5% HF aqueous solution for 2 min. Then, the HF-etched silicon nanograss were rinsed with deionized water twice

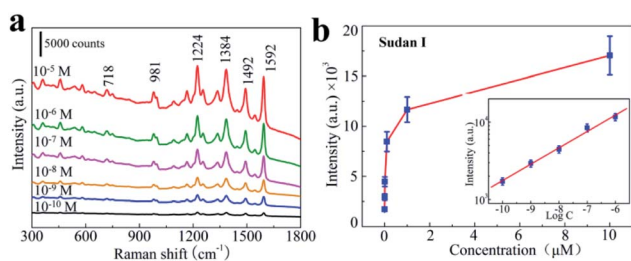


Fig. 8 (a) SERS spectra of Sudan I (incident power was 3.4 mW, data acquisition time was 10 s) at different concentrations; R6G Raman spectra; (b) plot of the peak intensity at  $1592\text{ cm}^{-1}$  as a function of Sudan I concentration (inset represents the linear relationship from  $10^{-10}$  M to  $10^{-6}$  M).



and immediately dipped into aqueous solutions of silver(i) nitrate shaking slightly for one minute. After the galvanic displacement reaction, the treated silicon nanograss were rinsed in deionized water and dried immediately by nitrogen blow and transferred into a vacuum chamber for further measurements.

The structure and morphology of the obtained silicon nanograss arrays were characterized X-Ray Diffraction (XRD, Bruker D8 ADVANCE), Scanning Electron Microscopy (SEM, Hitachi S4800) and Transmission Electron Microscopy (TEM, JEM-2100F). The surface state of the as-synthesized SERS substrates were analyzed by PHI-5400 X-ray photoelectron spectrometer (Perkin Elmer Inc) using an X-ray Al-K $\alpha$  source as a probe. An electron flood gun was used for all measurements to compensate charging, and the final spectra were calibrated to the adventitious carbon C1s peak at 284.6 eV.

### 3.4 SERS measurements

Using the as-prepared AgNPs decorated 3D bionic silicon nanograss arrays as SERS-active substrates, and a UNISOKU-TII (Tokyo Instruments) laser confocal Raman spectrometer was used to carry out the SERS measurements. R6G was chosen as a model analyte to investigate the performance of the 3D SERS substrate because it is extensively studied and well characterized by Raman spectroscopy. The SERS measurements were performed by dipped the substrates into 5 mL R6G solution of different concentrations (from  $10^{-6}$  M to  $10^{-14}$  M) for 3 hours. He-Ne laser of 632.8 nm at a spectral resolution of  $1\text{ cm}^{-1}$  and a spot size of  $1\text{ }\mu\text{m}$  was used for excitation. The laser power on the specimens was measured at 3.4 mW. The recorded spectra were raw data without any background subtraction or processing.

## 4. Conclusions

In this study, a AgNPs decorated 3D bionic silicon nanograss arrays SERS substrate was successfully fabricated, using the shrinkage of Au catalyst droplet in VLS growth and subsequent galvanic displacement decoration. The clean surfaces, high-density hot spots, and significant Raman signal amplification were obtained by adjustable reaction time and silver ion concentration. The optimal AgNPs decorated 3D bionic silicon nanograss arrays were able to detect R6G down to 0.1 pM, and the analytical enhancement factor (AEF) was estimated to be  $3.3 \times 10^7$ , which is obviously superior to that of pure silicon or silver nanostructures. Importantly, a remarkably low concentration of Sudan I down to 0.1 nM (much lower than those traditional analytical approaches) can readily be discriminated *via* the developed SERS substrates. Besides the SERS sensing, the substrates obtained in this study might be widely applied in many different fields, such as medical diagnosis, environment supervision, and catalytic conversion.

## Author contributions

Jian Huang and Li Wang are considered co-first authors, both authors contributed equally to this work. All authors declare no competing financial interests.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This research was funded by the Natural Science Basic Research Program of Shaanxi (No. 2019JQ-490), Scientific Research Program Funded by Shaanxi Provincial Education Department (No. 21JK0835), and the Innovation and Practice Ability Training Project for Postgraduates of Xi'an Shiyou University (No. YCS20211022).

## Notes and references

- X. Yao, S. Jiang, S. Luo, B.-W. Liu, T.-X. Huang, S. Hu, J. Zhu, X. Wang and B. Ren, *ACS Appl. Mater. Interfaces*, 2020, **12**, 36505–36512.
- Y. Wang, B. Yan and L. Chen, *Chem. Rev.*, 2013, **113**, 1391–1423.
- J. Li, L. Chen, T. Lou and Y. Wang, *ACS Appl. Mater. Interfaces*, 2011, **3**, 3936–3941.
- R. Mei, Y. Wang, Q. Yu, *et al.*, *ACS Appl. Mater. Interfaces*, 2020, **12**, 2059–2066.
- F. Xu, M. Xuan, Z. Ben, W. Shang and G. Ma, *Rev. Anal. Chem.*, 2021, **40**, 75–92.
- P. Wang, Y. Sun, X. Li, L. Wang, Y. Xu and G. Li, *Molecules*, 2021, **26**, 209.
- M. Muhammad and Q. Huang, *Talanta*, 2021, **227**, 122188.
- A. Kapara, V. Brunton, D. Graham and K. Faulds, *Chem. Sci.*, 2020, **11**, 5819–5829.
- G. Pavliuk, D. Pavlov, E. Mitsai, O. Vitrik, A. Mironenko, A. Zakharenko, S. A. Kulinich, S. Juodkazis, S. Bratskaya, A. Zhizhchenko and A. Kuchmizhak, *Nanomaterials*, 2020, **10**, 49.
- X. Jing, L. Chang, L. Shi, X. Liu, Y. Zhao and W. Zhang, *ACS Appl. Bio Mater.*, 2020, **3**, 2385–2391.
- L. G. Quagliano, B. Jusserand and D. J. Orani, *Raman Spectrosc.*, 1998, **29**, 721–724.
- T. Wang, Z. Zhang, F. Liao, Q. Cai, Y. Li, S.-T. Lee and M. Shao, *Sci. Rep.*, 2014, **4**, 1–8.
- A. Fularz, S. Almohammed and J. H. Rice, *ACS Appl. Nano Mater.*, 2020, **3**, 1666–1673.
- M. Jue, C. G. Pack, S. Oh, B. Paulson, K. Lee and J. K. Kim, *Materials*, 2020, **13**, 5321.
- P. Zhang, G. Liu, W. Xu, L. Meng, X. Wang, L. Shang, Y. Xiong, Q. Luo and S. Feng, *ACS Omega*, 2020, **5**, 31730–31737.
- J. Huang, Y. Sun, S. Huang, K. Yu, Q. Zhao, F. Peng, H. Yu, H. Wang and J. Yang, *J. Mater. Chem.*, 2011, **21**, 17930–17937.
- L. Sun, J. He, S. An, J. Zhang and D. Ren, *J. Mol. Struct.*, 2013, **1046**, 74–81.
- S. Zhu, C. Fan, J. Wang, J. He, E. Liang and M. Chao, *J. Colloid Interface Sci.*, 2015, **438**, 116–121.
- M. Muniz-Miranda, C. Gellini, E. Giorgetti and G. Margheri, *J. Colloid Interface Sci.*, 2017, **489**, 100–105.



## Paper

- 20 J. Cai, J. Huang, M. Ge, J. Iocozzia, Z. Lin, K. Zhang and Y. Lai, *Small*, 2017, **13**, 1604240.
- 21 S. Park, P. Yang, P. Corredor and M. J. Weaver, *J. Am. Chem. Soc.*, 2002, **124**, 2428–2429.
- 22 T. Hinamoto, S. Hotta, H. Sugimoto and M. Fujii, *Nano Lett.*, 2020, **20**, 7737–7743.
- 23 X. Wan, X. Chen, S. Yang, W. Ma, S. Li and K. Wei, *Mater. Lett.*, 2020, **268**, 127616.
- 24 S. Kuhn, P. Asenbaum, A. Kosloff, M. Sclafani, B. A. Stickler, S. Nimmrichter, K. Hornberger, O. Cheshnovsky, F. Patolsky and M. Arndt, *Nano Lett.*, 2015, **15**, 5604–5608.
- 25 M.-L. Zhang, X. Fan, H.-W. Zhou, M.-W. Shao, J. A. Zapien, N.-B. Wong and S.-T. Lee, *J. Phys. Chem. C*, 2010, **114**, 1969–1975.
- 26 B. Reddy G, R. Dadigala, R. Bandi, *et al.*, *RSC Adv.*, 2021, **11**, 5139–5148.
- 27 C. Fang, A. Agarwal, H. Ji, W. Y. Karen and L. Yobas, *Nanotechnology*, 2009, **20**, 405604.
- 28 B. Hua, W. Zhang, L. Chen, *et al.*, *J. Appl. Phys.*, 2012, **112**, 124301–124306.
- 29 R. Frederiksen, G. Tutuncuoglu, F. Matteini, *et al.*, *ACS Photon.*, 2017, **4**, 2235–2241.
- 30 S. Botti, R. Ciardi, R. Larciprete, A. Goldoni, L. Gregoratti, B. Kaulich and M. Kiskinova, *Chem. Phys. Lett.*, 2003, **371**, 394–400.
- 31 L. Wang, H. Li, J. Tian and X. Sun, *ACS Appl. Mater. Interfaces*, 2010, **2**, 2987–2991.
- 32 P. M. Ryan, L. Livadaru, G. A. DiLabio and R. A. Wolkow, *J. Am. Chem. Soc.*, 2012, **134**, 12054–12063.
- 33 S. R. Emory, W. E. Haskins and S. Nie, *J. Am. Chem. Soc.*, 1998, **120**, 8009–8010.
- 34 Y. Zhao, G. Chen, Y. Du, J. Xu, S. Wu, Y. Qu and Y. Zhu, *Nanoscale*, 2014, **6**, 13754–13760.
- 35 C. Di Anibal, M. S. Rodriguez and L. Albertengo, *Food Anal. Methods*, 2014, **7**, 1090–1096.
- 36 Q. Shao, R. Que, M. Shao, L. Cheng and S.-T. Lee, *Adv. Funct. Mater.*, 2012, **22**, 2067–2070.
- 37 X. Kong, K. Squire, X. Chong and A. X. Wang, *Food Control*, 2017, **79**, 258–265.
- 38 Q. Cao, X. Liu, K. Yuan, J. Yu, Q. Liu, J.-J. Delaunay and R. Che, *Appl. Catal., B*, 2017, **201**, 607–616.
- 39 B. Liu, P. Zhou, X. Liu, X. Sun, H. Li and M. Lin, *Food Bioprocess Technol.*, 2013, **6**, 710–718.
- 40 X. Lu, W. Liu, C. Chen, G. Jiang, X. Hu, H. Zhang and M. Zhong, *Laser Technol.*, 2021, **139**, 106969.

