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# Synthesis of holey graphene for advanced nanotechnological applications

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Holey or porous graphene, a structural derivative of graphene, has attracted immense attention due to its unique properties and potential applications in different branches of science and technology. In this review, the synthesis methods of holey or porous graphene/graphene oxide are systematically summarized and their potential applications in different areas are discussed. The process–structure–applications are explained, which helps relate the synthesis approaches to their corresponding key applications. The review paper is anticipated to benefit the readers in understanding the different synthesis methods of holey graphene, their key parameters to control the pore size distribution, advantages and limitations, and their potential applications in various fields.

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## 1. Introduction

Graphene is a remarkable material with excellent material properties such as high thermal and electrical conductivity, high electron mobility and Young's modulus.<sup>1–3</sup> However, the real-world or commercial applications of graphene have been limited partly due to its zero band gap nature, which is considered a bottleneck for its application in the semiconductor industry and partly due to the complexity in the production of high-quality graphene economically.<sup>4</sup> Heterostructure devices such as graphene–hBN and graphene–MoS<sub>2</sub>/WS<sub>2</sub>, made of graphene as one of the components, have been rising to make profound use of the properties of graphene.<sup>5–9</sup> Another development that recently came out in the field of graphene is holey/porous graphene, which is considered a unique structural derivative of graphene/GO. Fig. 1a is a chart that shows the increasing number of publications focused on holey graphene in the past few years. The trend indicates a growing interest in the material due to its unique properties and versatile nature, and potential application in different branches of science and technology. Holey graphene (HG) is defined as a graphene material with a large number of holes or pores on its basal plane. The pore sizes can vary from a few nm to few tens of nm to few microns in size and can be created by various methods and techniques. The schematic picture (Fig. 1b) shows a few graphene layers with arbitrarily distributed pores on their basal planes. HG sheets, when assembled to form a 3D porous material, is then called 'porous graphene';<sup>10,11</sup> thus,

HG can be considered as a sub-category of 'porous graphene'. However, to have a comprehensive understanding of the synthesis processes and due to their similar properties and structural nature, by HG, we refer to both types of materials in this article. As indicated in the schematic Fig. 1, the irregularly arranged pores on the sheets form 3D channels. These 3D channels make a path for ions and other molecules to move around. In graphene sheets without pores (graphite), any ion or molecule will have two degrees of freedom. On the other hand, due to the presence of pores, ions and molecules will have three degrees of freedom in the HG material. Thus, graphene sheets with holes have the advantage of efficient ion transport even when the sheets are stacked.

Another aspect of holey graphene material is that due to the lack of continuous C atoms, interlayer attractive forces, *i.e.*, the van der Waals interactions among the layers, become weak. This manifests itself in the increased interlayer distance among the sheets. The availability of a large number of pores and increased interlayer distance provide a boost to the ion movement and enhanced mass transport. This opens up a major application of holey/porous graphene in the area of water purification and desalination.<sup>12–16</sup>

The origin of the restacking or aggregation in graphite or multilayer graphene is due to van der Waals interactions among the graphene sheets/layers. This force is proportional to the overlapping surface area ( $S$ ) and inverse power to the interlayer distance. Previous calculations indicated a fourth inverse power dependence;<sup>17</sup> however, after semiempirical corrections made to the power law, the interaction shows a asymptotic third inverse power dependence.<sup>18</sup> The inverse power value in the van der Waals force could also depend on the geometry of the two surfaces (*e.g.*, parallel plate and sphere) as well as the interplanar distance between the two objects.<sup>19–21</sup> In porous or holey

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**Fig. 1** (a) The chart shows the number of publications focused on holey/porous graphene in the last ten years (data obtained from Scopus). (b) The schematic image shows the presence of nanoscale pores in a stack of graphene sheets. (c) The cross-sectional schematic view shows the interlayer channels formed in a stack of graphene sheets due to randomly distributed pores. Ions or molecules with sizes lower than the pore sizes could pass through the material (shown by the arrows).

graphene material, restacking could be avoided due to the lack of enough carbon atoms and hence the lack of sufficient van der Waals forces. The limited restacking and the presence of a high number of pores could result in higher ionic access to the surfaces and thus increased specific surface area.

In addition to the fascinating properties, as discussed above, it is also found that porous graphene could behave similar to a direct band gap semiconductor. A study reveals a band gap value of 2.35 eV for porous graphene.<sup>22</sup> However, the value might depend on the computational method. Du *et al.* predicted a value of 3.2 eV from their density functional theory (DFT) calculations.<sup>23</sup> Another study from first principles calculations estimates an optical band gap of 1.28 eV.<sup>24</sup> The band gap is likely to be tuned with the size distribution of the pores in the material. Denis *et al.* investigated the impact of different type vacancies such as single, double, and Stone–Wales, on the electronic properties of monolayer and bilayer graphene.<sup>25</sup> The DFT study revealed that the creation of a single vacancy in bilayer graphene could induce metallic properties in the material; Stone–Wales defects would not alter the electronic properties much; on the other hand, mono/bilayer graphene with double vacancies would behave as a semi-metallic. These studies reveal the possibility of tuning the electronic properties of graphene with different pore (vacancy) types and sizes. However, a systematic experimental investigation is required to properly tune the electronic properties and further utilization in designing optoelectronic devices.

Another aspect of pores in HG is the high degree of reactivity of the unsaturated pores due to which the pore edges could be naturally saturated. For example, an unsaturated pore can chemisorb gas molecules and could be used for gas sensing

applications.<sup>26</sup> A theoretical investigation on the reactivity of different types of defects was carried out.<sup>27</sup> It was revealed that the reactivity would highly depend on the functional groups attached and the number of such attached groups. The study indicated that the reactivity would be higher for the lower binding energy of the functional group linked to the graphene sites.

Pores can also be created on sheets of other structural derivatives of graphene such as graphene oxide (GO) and reduced graphene oxide (rGO). This can also result in the formation of holey GO/rGO (HGO or HrGO). Similar to the case of HG, the presence of pores in GO/rGO can modify the intrinsic properties of GO/rGO and in many ways, HG or HGO/HrGO would have similar properties that could be useful in certain fields. Both GO and rGO have a large number functional groups that could attach to the carbon atoms in graphene. GO is a material with a high amount of oxygen attached to it in the form of carbonyl and hydroxyl groups and can be considered as the oxidized form of graphene. On the other hand, reduced graphene oxide, which is obtained after the reduction of GO, contains lesser number of functional groups as compared to GO. Thus, in contrast to pure HG materials that would not have any element other than carbon, HGO/HrGO would have a certain number of functional groups attached to the sheets. The structural difference among graphene, GO, and rGO can be understood from the schematic presented in Fig. 2. As indicated in the figure, graphene is a semi-metal with zero band gap. On the other hand, GO has a large number of carboxyl groups due to which it shows high band gap and rGO could behave like a semiconductor with variable band gap, depending on the amount of the attached functional groups. Thus, as compared to GO, which is an insulator in nature due to the presence of a large number of carboxyl groups, rGO would have some conductivity.





Fig. 2 Schematic figure and the corresponding energy band diagram of graphene, GO, and rGO.<sup>32</sup>

The approximate interlayer distance in a few layers of graphene sheet is about 0.33 nm. In the case of GO, the oxygen containing functional groups could increase the thickness of single layer GO to 0.55 nm (theoretical value).<sup>28</sup> However, in a multilayer GO sample, the interlayer distance could be 0.80–1.09 nm, which could vary largely depending on the amount of the attached functional groups.<sup>29</sup> The interlayer distance in rGO is higher than the interlayer distance in graphene but lower than the interlayer distance of GO.<sup>30</sup> The conductivity and the interlayer distance could be tuned by controlling the functional groups on the GO/rGO sample or the reduction process.<sup>31</sup> These properties are summarized in Table 1.

In addition to the properties mentioned above, GO or rGO can also provide high specific surface area; this property could be used in energy storage applications.<sup>35,36</sup> The properties of GO or rGO can be further improved or modified by creating holes in the materials. Due to the presence of holes, and hence the increased number of edges, the catalytic property of holey GO (HGO) can be relatively higher than that of GO/rGO. Similar to

HG, HGO/HrGO have also shown potential applications in the areas of energy storage, gas sensing, *etc.*<sup>37–41</sup>

HG, along with HGO/HrGO, has shown various interesting properties and various methods have been developed and implemented to synthesize or modify the materials for numerous applications. Due to the similar properties and applications of HG and HGO or HrGO by the term, HG we in general refer to HGO or HrGO along with holey/porous graphene. In this review, the main synthesis processes of HG and their major applications in different areas are discussed. HG is an emerging material; the synthesis and application of HG has been continuously evolving and various synthesis routes have been proposed in the recent few years as well as demonstrated to synthesize high quality holey/porous graphene. In recent years, articles focusing on different aspects of the material have been reported.<sup>42–45</sup> This review presents a systematic understanding of different synthesis routes to obtain holey graphene, its properties, and key applications in different fields. The article is expected to benefit the community for understanding

Table 1 Different properties of graphene, GO, and rGO.<sup>29,33,34</sup> Reprinted and adapted with permission.<sup>12</sup> Copyright 2015, Royal Society of Chemistry

Properties	Graphene	Graphene oxide	Reduced graphene oxide
Synthesis	Chemical vapor deposition, thermal decomposition of silicon carbide (SiC), graphite exfoliation	Oxidation and exfoliation of graphite	Reduction of graphene oxide
C : O ratio	No oxygen	2–4	8–246
Interlayer distance (nm)	0.33	0.80–1.09	0.35–0.45
Young's modulus (TPa)	1	0.2	0.25
Electron mobility ( $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ )	10 000–50 000	Insulator	0.05–200
Production cost	High	Low	Low



the synthesis methods of holey graphene, key parameters to control the pore size distribution, advantages and limitations of the methods, and their demonstrated applications in different fields. The article also presents a timely assessment to understand the current progress along with the recent trends and future opportunities.

## 2. Synthesis of holey graphene

Various methods have been developed by different research groups over the last few years in order to create or introduce holes on a graphene sheet or in a bulk graphene material. These methods can be categorized as physical, thermal, and chemical (Fig. 3). Energetic electron and ion beam can be used to remove carbon atoms from graphene sheets. This physical process is based on the energy and momentum transfer process from the charged particles to the carbon chains in the graphene sheet. An energetic electron or an ion can be generated in a typical scanning electron microscope (SEM), focused ion beam (FIB), or ion accelerator system. Pores can also be generated using chemical etching processes. The chemical methods can be further categorized into subcategories based on the approach. For example, chemical agents such as  $\text{H}_2\text{O}_2$  and  $\text{HNO}_3$  can be directly used to create pores in graphene. Nanoparticles are also used as a catalyst for guided etching on the graphene sheets. Also, nanolithographic chemical methods are developed for pore creation, also known as the template-assisted approach. Template-based processes can provide the periodic formation of holes. Different approaches could produce pores of different sizes and retain certain advantages and disadvantages. The pores in graphene can be classified as microporous (<2 nm), mesoporous (2–50 nm), and macroporous (>50 nm) according to the definition by International Union of Pure and Applied Chemistry (IUPAC).<sup>46,47</sup> In the following section, these synthesis methods are discussed in detail.

### 2.1 Physical: electron or ion beam irradiation

Micro/nano holes on a graphene sheet can be made by removing certain sets of carbon atoms. This can be achieved by passing energetic electrons or ion particles through the graphene sheet. The energetic particles transfer their momentum to the bonded carbon atoms. If the momentum transfer or

transferred energy is more than the binding energy of the  $\text{sp}^2$  bonded carbon atoms, then those carbon atoms can be removed from the site.

The fabrication of precise nanopores with nanometric precision on graphene sheets was demonstrated by Fischbein and co-workers using energetic electron beam in a 200 kV transmission electron microscopy (TEM) system.<sup>48</sup> The nanopores created in this way are shown in the images (Fig. 4). In addition to the nanoholes, nanobridges and nanogaps were also created using the energetic electron beam. Usually, the use of energetic electron beam poses an adverse effect on the crystallinity of the surrounding graphene materials and the graphene material could be amorphized. In many situations, this change in the crystallinity is highly unwanted as it is likely to change the optoelectronic properties of the material. However, this issue was found to be resolved when the nanopores were fabricated at temperatures above 600 °C.<sup>49</sup>

A low energetic electron beam has also been shown to be an effective approach in creating pores on graphene sheets. In the work by Fox *et al.*, <10 keV focused electron beam in an SEM system was used to create pores in the presence of nitrogen gas.<sup>50</sup>

The stability of the holes formed by energetic electrons was studied by Girit *et al.* using a TEM system.<sup>51</sup> The study revealed the complex behavior of the carbon atoms at the edge. The edge reconfiguration process and hence the change in the shape of the hole and stability of the zigzag configuration over the armchair type were observed. Schneider *et al.* further demonstrated the use of such pores in DNA relocation.<sup>52</sup>

On the other hand, Russo *et al.* combined both ion and electron beam to create nanopores.<sup>53</sup> Initially, 3 keV Ar ions



Fig. 3 Different synthesis methods to create pores in graphene and GO/rGO materials.

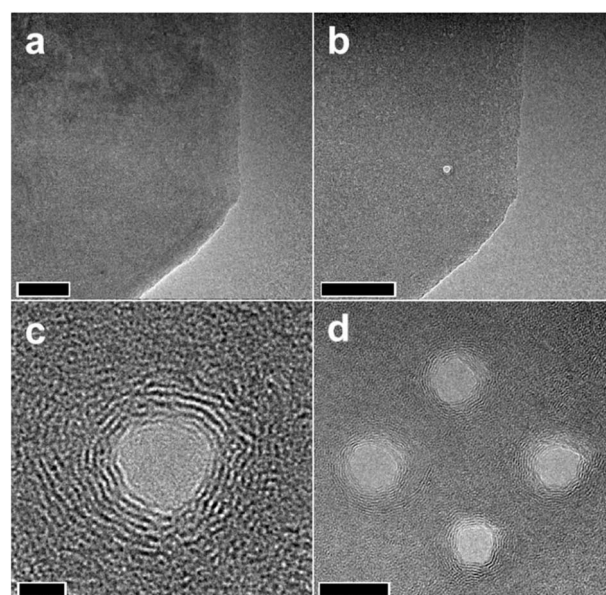


Fig. 4 TEM images of a graphene sheet (a) before and (b) after an energetic electron beam makes a nanopore. (c) Higher magnification image of the nanopore. (d) Multiple nanopores are made close to each other. Scale bars are 50, 50, 2, and 10 nm, respectively. Reprinted and adapted with permission.<sup>48</sup> Copyright 2008, AIP Publishing.



were used at low temperature (148 K) to create defect nucleation centers on a single-layer graphene sheet. The sample was then warmed to room temperature (RT) and irradiated with an 80 keV electron beam to create pores in the nanometer scale.

Similar to the electron beam, a focused ion beam (FIB) can also be used to create pores on a graphene sheet. As compared to the case of the electron beam, an ion beam can be very effective in creating holes as ions are much heavier than electrons; thus, momentum transfer from the ions to the bonded carbon atoms is easier. Wang *et al.* showed the fabrication of nanometric holes on a single-layer graphene sheet using 30 kV Ga ion beam.<sup>54</sup> The obtained pore size could be of few tens of nm and pores with an elliptical shape could be made as well. Similarly, Celebi *et al.* used FIB to create pores of the size 10 nm to 1 micron on double layer graphene.<sup>55</sup> The pores could be used to study the transport phenomenon of liquid, gas, *etc.* Similar works were carried out by O'Hern *et al.* to create pores of size  $0.40 \pm 0.24$  nm.<sup>56</sup> The fabrication of tunable holes in the range of 1–4 nm diameter was demonstrated by Vazquez *et al.* using swift heavy ions.<sup>57</sup> Swift heavy ions generally refer to the heavy mass with high energy (>MeV) particles and can be generated in a particle accelerator.<sup>58</sup> Such energetic particles can produce

tracks on their path in a material and when mono or few layers of graphene is used, holes can be formed.

In another study conducted by Nebogatikova *et al.*, the formation of 20–60 nm size nanopores was demonstrated using Xe swift heavy ions.<sup>59</sup> The sizes of the pores were found to be independent of the energy and dose of the ions, and the number of pores can be increased with the ion dose.

An interesting simulation study carried out by Li *et al.* on the fabrication of nanopores on graphene sheet by ion beam suggested the use of Au ions for creating perfect nanopores with smooth edges.<sup>60</sup> The study also unveiled that the use of higher ion beam energy leads to lower damage. When carbon ions are used, it is indicated that with energy more than 5 keV, a complete nanopore cannot be formed; however, this relation vanishes when the incident ion beam mass increases.

Pore fabrication approaches using electron and ion beam are summarized in Table 2. Even though energetic electron and ion beams have been very effective in fabricating porous graphene sheets in a controllable and precise way, the technique has limitations in the fabrication of large-scale porous graphene sheets. Thus, this method is not a suitable approach for the mass scale synthesis of holey graphene material economically.

Table 2 Electron and ion beam-assisted pore generation on graphene

Beam type	Energy regime	Beam energy (and ion type)	Mechanism/method	Pore size	Application	Ref.
Electron beam	Low energy (SEM)	<10 keV	Electron beam was focused on the graphene sheet in the presence of N <sub>2</sub> gas. N <sub>2</sub> was ionized, which then reacts with C atoms to form a gaseous product	<10 nm	—	Fox <i>et al.</i> <sup>50</sup>
	High energy (TEM)	200 keV	Focused electron beam-induced ablation	3.5 nm	—	Fischbein <i>et al.</i> <sup>48</sup>
		300 keV	Electron beam-induced knock out and temperature-assisted self-repair of the amorphized region	~(1–10) nm	—	Song <i>et al.</i> <sup>49</sup>
		200 keV	Pores are created on a graphene sheet suspended on an SiN <sub>x</sub> membrane	2–40 nm	DNA translocation	Schneider <i>et al.</i> <sup>52</sup>
		80 keV	3 keV Ar ions were irradiated at low temperature to create defects, followed by electron beam irradiation at RT	0.3 nm	—	Russo <i>et al.</i> <sup>53</sup>
Ion beam	Low energy (FIB/ion sputtering system)	~keV Ga, He	Focused ion beam-induced drilling	<10 nm to 1 μm	Permeation membrane	Celebi <i>et al.</i> <sup>55</sup>
		5–30 keV: Ga 10–30 keV: He 8 keV: Ga	Ion beam milling	<10 nm	—	Buchheim <i>et al.</i> <sup>62</sup> O'Hern <i>et al.</i> <sup>56</sup>
	High energy (swift heavy ion)	MeV range: Xe, I, O, Si, C, Au	Ion beam etching	0.40 ± 0.24 nm	—	Vazquez <i>et al.</i> <sup>57</sup>
		26–167 MeV: Xe	The SHIs created structural modifications with cylindrical symmetry ('tracks') along the ion path within the target material	1–4 nm	—	
		Energy transfer <i>via</i> the electronic stopping process (ionization loss)	20–60 nm	—	Nebogatikova <i>et al.</i> <sup>59</sup>	



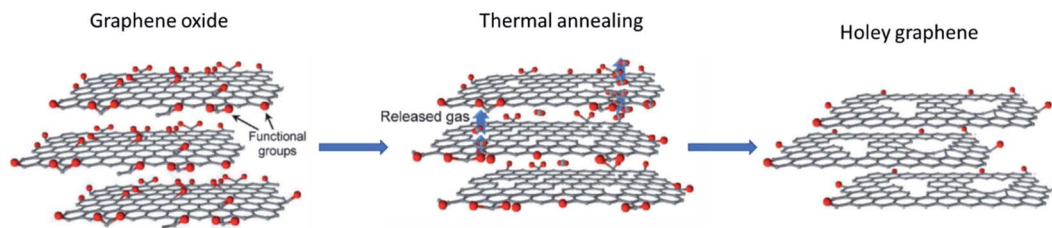


Fig. 5 Schematic representation of the formation of holey graphene from graphene oxide using rapid thermal annealing. Reprinted and adapted with permission.<sup>66</sup> Copyright 2015, John Wiley and Sons.

## 2.2 Thermal: annealing method

It has been reported that upon heat treatment of GO or rGO, the process could result in the formation of HG.<sup>63–65</sup> Structurally, both GO and rGO are similar as they are both derivatives of graphite or graphene and both have functional groups in the form of hydroxyl, oxygen, or carboxyl groups.

**2.2.1 Rapid thermal annealing (RTA).** When GO or rGO is heated or annealed at a desired temperature, the functional groups are decomposed and various gases such as CO<sub>2</sub> are formed.<sup>66</sup> If a sufficient amount of CO<sub>2</sub> is formed due to annealing, pressure would build up among the layers. This could lead to forced liberation of the CO<sub>2</sub> gas molecules from the layers. The high pressure could punch through the graphene sheets and several holes can be created in this way. The process is schematically shown in Fig. 5. However, the phenomenon was observed only when the heating rate was relatively high, for *e.g.*, 100 °C s<sup>-1</sup>. The study also reveals that pore formation is associated with the heating rate and higher heating generally leads to higher pore size and pore density.<sup>66</sup>

A similar low-cost and simple method based on rapid thermal annealing was demonstrated by Peng *et al.*<sup>63</sup> In their work, GO powder was taken as the starting material and rapidly annealed at a temperature as low as 300 °C. Initially, the quartz tube furnace was heated to 300 °C and kept for 30 min. Thereafter, a long-handled stainless-steel lab spoon containing the GO powder was inserted. The estimated heating rate was 100 °C s<sup>-1</sup> or 6000 °C min<sup>-1</sup>. Holes with sizes in the range of 10–250 nm were created. The number of

holes was controlled with the ramping rate of temperature increase. The grown HG nanosheets have shown promising results in supercapacitor applications. The specific capacitance at all the current densities was found to increase with the number of holes as created by raising the heating rate. The supercapacitor made with the holey graphene nanosheets showed a specific capacitance of 211 F g<sup>-1</sup> at 0.5 A g<sup>-1</sup>, which was 49% higher than that of the supercapacitor made with the non-holey counterpart.

Wu *et al.* considered a similar approach to gain a higher ramping rate for the RTA process.<sup>64</sup> First, a tube furnace filled with Ar atmosphere was preheated to 1100 °C and kept there. In the next step, a long metal tool was used to place the GO powder in the central region of the furnace. This allows a high ramping rate of 18 000 °C min<sup>-1</sup>. This ultra-high ramping rate produces a large amount of CO<sub>2</sub> gas in between the neighboring layers in a short span of time. It leads to the generation of huge pressure. This results in the piercing of the graphene sheets at their weak spots thus forming holes. HG was synthesized in this way and subsequently applied in lithium-ion battery (LIB).

Xing *et al.* used anthracite (raw coal) and graphitized it using a high temperature furnace.<sup>67</sup> Later, Hummers' method was utilized to obtain coal-based GO (CGO) from the graphitized sample. The CGO sample was then placed in a crucible and heated rapidly at 900 °C for 10 min in N<sub>2</sub> flow to generate pores in the sample. Corrugated nanosheets with micro-meso-macroporous structure were obtained for LIB application as an anode material. Table 3 summarizes the different rapid thermal

Table 3 Creating pores using the rapid thermal annealing method

Annealing type	Starting material	Temp. (°C); time	Gas atmosphere	Ramping rate	Final material	Pore size (nm)	SSA (m <sup>2</sup> g <sup>-1</sup> )	Application	Ref.
Rapid	GO powder	300	Air	100 °C s <sup>-1</sup>	HGNS	10–250	592	Supercapacitor	Peng <i>et al.</i> <sup>63</sup>
	GO	300	—	> 100 °C s <sup>-1</sup>	HGNS	50–100	750	Supercapacitor	Yang <i>et al.</i> <sup>66</sup>
	GO	1000	—	—	—	—	800	—	—
	GO	1100, 1 h	Ar	18 000 °C min <sup>-1</sup>	HG	5–500	779	LIB	Wu <i>et al.</i> <sup>64</sup>
	GO	400 °C, 10–60 min, 900 °C, 2 h	Air	—	HG	—	733.5 (10 min–900)	—	Zhou <i>et al.</i> <sup>68</sup>
	Coal-GO (anthracite)	900, 10 min	Nitrogen	—	Porous graphene	Micro-meso-macro	640	LIB	Xing <i>et al.</i> <sup>67</sup>
	GO	300	Ar	>100 °C s <sup>-1</sup>	HGNS	50–300	750	NaIB	Luo <i>et al.</i> <sup>69</sup>
	GO	1100	—	—	—	—	800	—	—
	GO powder	1100, 1 h	Ar	18 000 °C min <sup>-1</sup>	HG	5–300	780	LIB	Huang <i>et al.</i> <sup>70</sup>



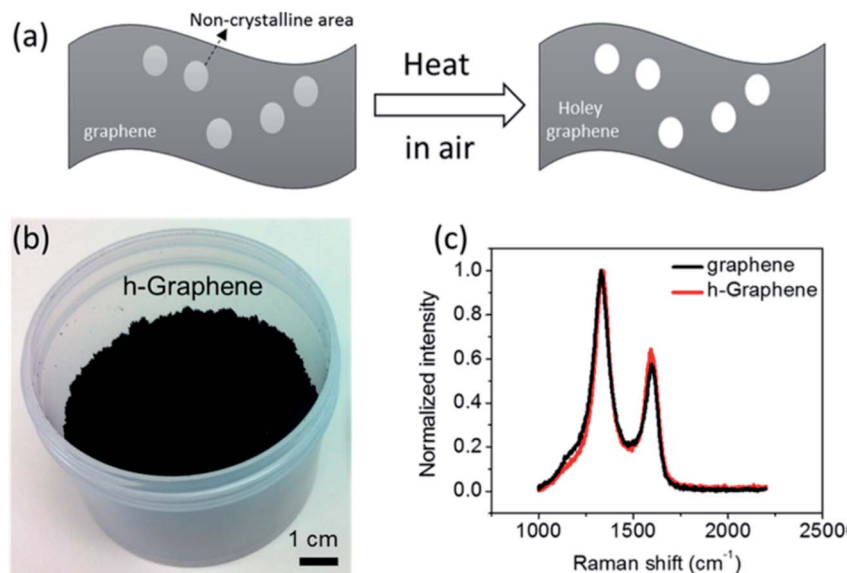


Fig. 6 (a) Schematic of holey graphene synthesis from graphene sheets, (b) as-synthesized holey graphene powder in bulk, and (c) Raman spectra obtained from holey graphene and the starting graphene material. Reprinted with permission.<sup>71</sup> Copyright 2014, American Chemical Society.

annealing treatments reported in the literature and the obtained pore sizes.

**2.2.2 Gradual thermal annealing (GTA).** In a separate work, gradual thermal annealing was implemented to create holey graphene<sup>11,71,72</sup> from graphene powder. It was demonstrated that the direct heating of graphene sheets in air at controlled temperature and time (with a ramping rate of  $\sim 10\text{ }^{\circ}\text{C min}^{-1}$  and held isothermally at a set temperature of  $<500\text{ }^{\circ}\text{C}$ ) can result in the formation of holey graphene. The defects that are present on the basal plane (referred to as the non-crystalline area) of the graphene sheet were found to react with oxygen in hot air ( $<500\text{ }^{\circ}\text{C}$ ) and resulted in the formation of nanometric holes on the graphene planes. The hole formation process is schematically shown in Fig. 6. This synthesis process does not use any specific chemical or catalyst and is found to be relatively robust and can be scaled up easily. Fig. 6b shows the obtained holey graphene material after the synthesis. The Raman spectrum in Fig. 6c indicates a relative decrease in the D to G ratio, which is thought to be because of the removal of the  $\text{sp}^3$  carbons from the defective regions (oxidative removal). The HG material synthesized in this way was evaluated for electrochemical properties and tested as an electrode in ultracapacitors, supercapacitor, and also experimented for  $\text{LiCO}_2$  battery.<sup>11,71,73</sup>

Following the same approach, Xu *et al.* synthesized HG from graphene material, after which N-doped HG was prepared by annealing graphene and HG at  $500\text{ }^{\circ}\text{C}$  for 5 h in a tube furnace in the presence of  $\text{NH}_3/\text{Ar}$  gas glow.<sup>74</sup> The doped HG was demonstrated to be an effective anode material for LIB.

A nanomanufacturing method was introduced by Chen *et al.* through the thermal annealing of commercially available few-layer graphene powder.<sup>75</sup> The powder was placed in a quartz boat and heated to  $\sim 710\text{ K}$  with a  $10\text{ K min}^{-1}$  heating rate and kept in an open-ended tube furnace for 10 h. The process gives

HG powder with an average pore size of 15 nm. The prepared holes could be repaired and closed if treated at a high temperature (2700 K).

Su *et al.* demonstrated a facile method to prepare HG nanoscrolls from GO nanosheets.<sup>65</sup> Schematically, the process is shown in Fig. 7. As the first step, large-size GO sheets were cold quenched in liquid nitrogen to synthesize the GO nanoscrolls. After that, GONSs were thermally annealed at  $500\text{ }^{\circ}\text{C}$  for 1 h in an  $\text{NH}_3$  environment to synthesize nitrogen-doped graphene nanosheets (NGNSs). The NGNSs were further annealed in air with an open-ended tube furnace at  $360\text{ }^{\circ}\text{C}$  for 5 h to obtain nitrogen-doped holey graphene nanosheets (NHGNSs). The NGNSs materials demonstrate promising results as electrode materials for supercapacitors and other electrochemical energy storage devices.

Xing *et al.* synthesized HG nanosheets (HGNSs) from oxidized graphite powder using a low temperature ( $300\text{ }^{\circ}\text{C}$ ) thermal treatment method.<sup>12</sup> Oxidized graphite powder was

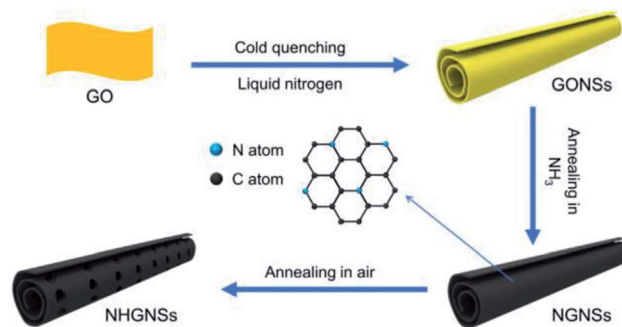


Fig. 7 Schematic picture showing the steps to prepare HG nanoscrolls, as demonstrated by Su *et al.* Reprinted with permission.<sup>65</sup> Copyright 2021, Elsevier.



Table 4 Pore generation using the gradual (normal) thermal annealing method

Annealing type	Starting material	Temp. (°C); time	Gas atmosphere	Ramping rate	Final material	Pore size (nm, avg.)	SSA (m <sup>2</sup> g <sup>-1</sup> )	Application	Ref.
Normal/ GTA	Graphene powder	~436.85 °C (710 K); 10 h	Air	10 K min <sup>-1</sup>	HG powder	15	—	—	Chen <i>et al.</i> <sup>75</sup>
	Graphene powder	<500 (430–480); 3 h, 10 h	Air	~10 °C min <sup>-1</sup>	HG	<10	658 (480 °C/10 h)	Ultracapacitor	Han <i>et al.</i> <sup>71</sup>
	Graphene powder	430, 10 h	Air	~10 °C min <sup>-1</sup>	HG	—	—	Li–CO <sub>2</sub> battery (HG-quantum dot composite)	Jin <i>et al.</i> <sup>72</sup>
	Graphene powder	430, 10 h	Air	10 °C min <sup>-1</sup>	HG	5–15	~600–700	Supercapacitor	Walsh <i>et al.</i> <sup>11</sup>
	Graphene	430; 3 h	Air	~10 °C min <sup>-1</sup>	HG	—	398.9	LIB	Xu <i>et al.</i> <sup>74</sup>
	Gr and HG	500; 5 h	NH <sub>3</sub> /Ar	—	N-Doped HG	—	594.6	—	—
	GO	360; 5 h	Air	5 °C min <sup>-1</sup>	NHGNSs	—	—	Supercapacitor	Su <i>et al.</i> <sup>65</sup>
	Oxidized graphite powder	300, 30 min	Air	—	HGNSs	1.4–16.3	1053.3	Water cleaning	Xing <i>et al.</i> <sup>12</sup>
	Graphite	800–1100, 1 h	Ar/H <sub>2</sub> O	—	HG	~5 nm (mesopore)–2 μm (micro)	4.4 (1000 °C)	LIB	Xiao <i>et al.</i> <sup>76</sup>

prepared using the modified brodie method and the prepared powder was put in a glass beaker to start the annealing process in air for 30 min. A high specific surface area (SSA) of 1053.3 m<sup>2</sup> g<sup>-1</sup> was obtained, which exhibits excellent properties in the absorption of oils, solvents, and dyes.

Xiao *et al.* described a low-cost, large-scale production method of HG from commercially available graphite powder.<sup>76</sup> The powder was placed in an alumina crucible and placed inside a furnace. The furnace was gradually heated up to the temperature range of 800–1100 °C and kept there for 1 h under the flow of the Ar/H<sub>2</sub>O gas mixture. H<sub>2</sub>O steam acts as a mild oxidative reagent in the process. Well-defined hexagonal hole structures of few microns in size and mesopores were obtained. The material was tested as the anode in LIB and showed promising results.

Table 4 summarizes the works of the GTA method. Holey/porous graphene synthesis *via* RTA process has been found to

be faster for HG fabrication than the GTA process, which usually takes time. Nevertheless, both types of annealing processes can synthesize HG materials, which showed promising results in energy storage applications.

### 2.3 Chemical: nanoparticle-assisted oxidation (catalytic oxidation)

In the work carried out by Lin *et al.*, a simple chemical process based on the catalytic property of silver (Ag) was utilized to create holes on graphene sheets.<sup>77</sup> Initially, Ag nanoparticles were deposited on the graphene sheets, and then the Ag coated samples were thermally annealed in air. The carbon atoms in contact with the Ag nanoparticles are oxidized and CO or CO<sub>2</sub> are formed. The process results in the formation of holes with a size that can be controlled by controlling the size of the Ag nanoparticles that are being used. The steps are shown in the schematic presented in Fig. 8. After the oxidation process, the

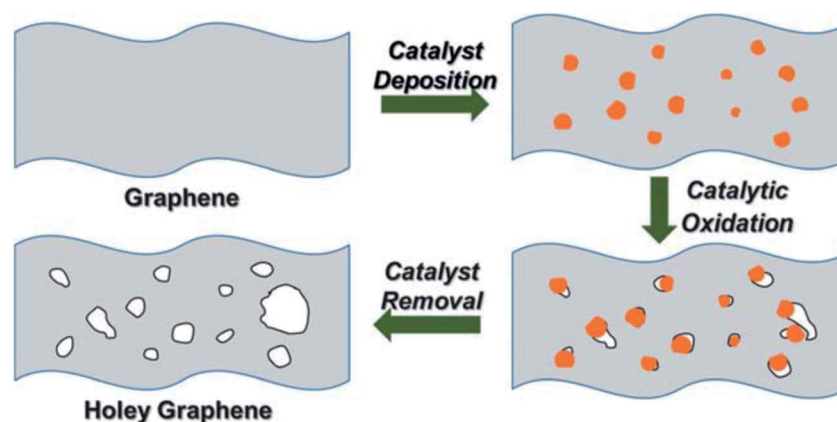


Fig. 8 The 3-step process to prepare holey graphene (HG) using Ag nanoparticles as discussed by Lin *et al.* Reprinted with permission.<sup>77</sup> Copyright 2013, Royal Society of Chemistry.







Fig. 9 Electron microscopy images of a holey graphene sample prepared using Ag nanoparticles as the catalyst. Reprinted with permission.<sup>77</sup> Copyright 2013, Royal Society of Chemistry.

Ag nanoparticles can be removed using nitric acid, followed by extensive washing with water and subsequent drying. The electron microscopy images of a prepared holey graphene sample utilizing this approach are shown in Fig. 9. The dark spots (SEM image: Fig. 9a) and the bright ones (TEM image: Fig. 9b) correspond to the holes formed during the process.

Alsharaeh *et al.* presented a different approach to obtain Ag-decorated GO using microwave heating.<sup>78</sup> Initially, GO was mixed with AgNO<sub>3</sub>, then sonicated and stirred. Thereafter, the solution was put inside a conventional microwave to obtain Ag-decorated rGO sheets. The Ag-rGO sample was then kept in a furnace at 300 °C for 2 h. Subsequently, the sample was refluxed with dilute nitric acid and microwave treated to remove the Ag particles. After washing in water and drying, HrGO was obtained. The obtained material was used as an anode in LIB.

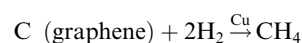
A chemical patterning method was proposed by Seo *et al.* using Fe<sup>III</sup> porphyrin covalent organic framework (COF) as a surface catalyst and template.<sup>79</sup> HOPG material was taken as a starting material for catalytic oxidation. The oxidation process was carried out with H<sub>2</sub>O<sub>2</sub> and NaOCl. Using this method, regular holes were created on the HOPG sample.

In a recent work performed by Wang *et al.*, ZnO nanoparticles were used with GO to create holes on the GO sheets.<sup>80</sup> A

solution was prepared with GO and ZnO to prepare a layer-by-layer pattern of GO–ZnO, as schematically shown in Fig. 10. This composite material was then calcined at a high temperature. During the calcination process, graphene was formed and the ZnO nanoparticles etch the adjacent graphene sheets, thus creating holes. Zn metal is produced as a side product, which could be removed by heating the material at 900 °C.

A similar approach was implemented by Du *et al.*<sup>81</sup> In their work, Fe<sub>2</sub>O<sub>3</sub> nanoparticles were formed on graphene sheets by implementing a simple hydrothermal reaction between FeCl<sub>3</sub> and GO. On further annealing, the Fe<sub>2</sub>O<sub>3</sub> nanoparticles can etch the surrounding graphene sheets and create holes in this way. The material was used in Li ion storage.

A Cu nanoparticle-assisted oxidation method was described by Dutta *et al.*<sup>82</sup> Initially, a Cu–graphene solution was prepared by mixing the Cu<sup>2+</sup> salt with the graphene material. The mixture was then annealed in two ways. First, rapid thermal annealing (RTA) in H<sub>2</sub> environment for 15 min and then gradual thermal annealing (GTA) at 900 °C for 15 min (with H<sub>2</sub> upon onset and until the end at 900 °C) was done. It was found that Cu could be utilized as a catalyst for the reaction of sp<sup>2</sup> carbon with H<sub>2</sub> to produce CH<sub>4</sub>, as shown in the equation below.



Following the annealing process, Cu nanoparticles can be removed using dilute nitric acid. The prepared Cu–graphene mixture with densely decorated Cu NPs is shown in Fig. 11a. Cu NPs have a mean particle size of 23.5 nm. Fig. 11b shows the HG sample after Cu NPs removal and the inset indicates a high degree of crystallinity of HG. Fig. 11c is a high-resolution bright field TEM image of a uniform HG (u-HG) obtained by RTA at 900 °C. It has a mean pore size of 24.2 nm, which can be correlated to the mean Cu NP size of 23.5 nm. The hierarchical presence of micro-, meso-, and macroscale pores in the few layers of graphene was observed. The TEM image in Fig. 11d shows a range of hierarchical pores observed at different layers

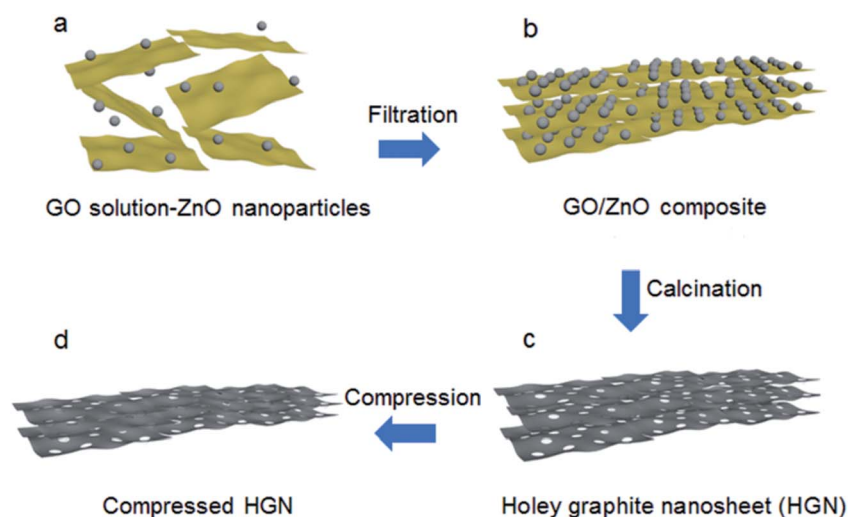
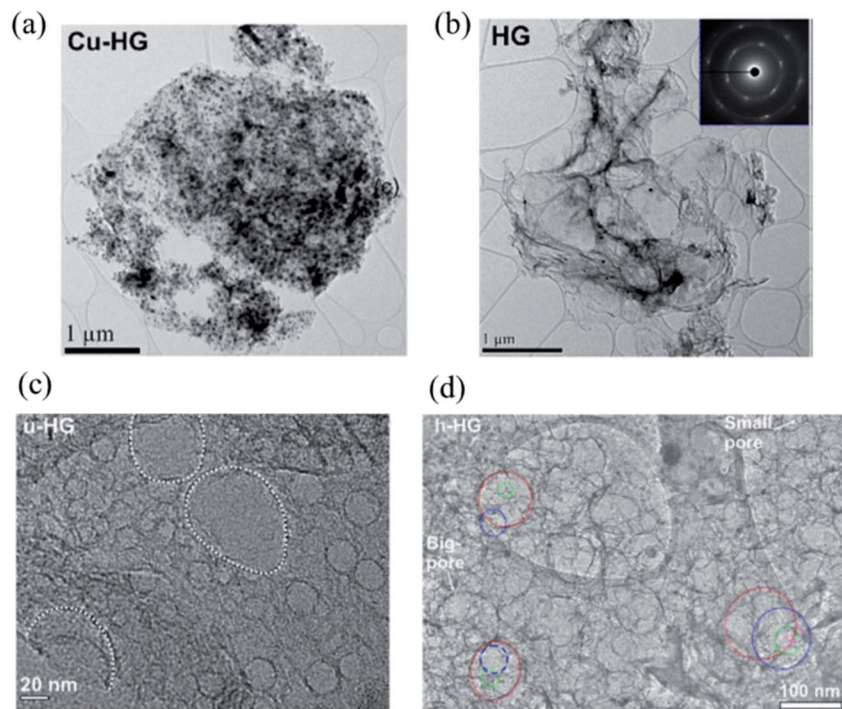


Fig. 10 The synthesis approach to produce holey graphene sheets from GO using ZnO nanoparticles. Reprinted with permission.<sup>80</sup> Copyright 2018, Royal Society of Chemistry.





**Fig. 11** (a) TEM image of the Cu–graphene composite. (b) TEM image of the graphene sheet after the removal of the Cu nanoparticles; selected area electron diffraction (SAED) pattern of HG is shown in the inset. (c) Higher-magnification TEM image of uniform HG (u-HG) showing a monodispersed distribution of pores and (d) the TEM image of hierarchical h-HG (pores on different layers of the h-HG; red, top layer; blue, 2nd layer; green, 3rd layer; magenta, 4th layer). Reprinted and adapted with permission.<sup>82</sup> Copyright 2019, American Chemical Society.

stacked one above the other. The tailoring of HG with optimized porosity results in high areal capacitance and excellent cycling stability, indicating that HG is a promising material for energy storage applications.

Table 5 summarizes the use of different types of nanoparticles for the synthesis of HG. Nanoparticle-assisted oxidation and pore creation are exciting approaches due to the porosity tailoring capability of the method through the nanoparticle size distribution. However, the technique may not be a suitable approach for the mass-scale production of HG.

#### 2.4 Chemical: chemical etching method

HG can be obtained from GO by means of wet chemical methods as well. Zhao *et al.* prepared a mixed solution of GO and HNO<sub>3</sub>.<sup>83</sup> The solution was then sonicated for 1 h. Sonication helps HNO<sub>3</sub> to react with the unsaturated carbon atoms at the damage sites and the edge sites of GO. The process results in the removal of carbon atoms from the GO sheet and accordingly, the formation of nanopores. Pores in the size range of 7–600 nm could be obtained in different holey GO (HGO) samples. Reduced HGO was prepared after thermally reducing the porous samples (Fig. 12) and paper electrodes were prepared, which exhibited enhanced Li-ion storage capacity and transport properties.<sup>83</sup>

A nanomesh of graphene (GNM) was prepared by Wang *et al.* by refluxing rGO sheets in concentrated HNO<sub>3</sub> solution for 4, 6, 9, or 11 hours.<sup>84</sup> The porous structures of GNMs could be controlled with the reaction time and the size of the nanopores

could be modulated from a few to hundreds of nanometers. The use of HNO<sub>3</sub> was also reported in another study by Chang *et al.* who prepared graphene electrodes using self-assembled 3D graphene made of graphene flakes with nanopores (~2.5 nm) for high-performance supercapacitor application.<sup>85</sup> A three-step approach was implemented, as shown in Fig. 13, to obtain a 3D assembly of micro/nanoporous GO. First, the GO solution was mixed with HNO<sub>3</sub>. HNO<sub>3</sub> then reacts with the defect sites on the edges of GO. The process results in the removal of carbon atoms from the GO flakes and holey GO is formed. In the next step, freeze-drying was used to obtain the 3D graphene monolith and lastly, thermal reduction was carried out to get 3D porous graphene.

Similarly, Chowdhury *et al.* prepared HGO by sonicating a mixture of aqueous GO dispersed solution and HNO<sub>3</sub>.<sup>86</sup> The as-prepared HGO was then annealed under N<sub>2</sub> gas at 500 °C for 30 min to obtain a holey graphene framework. The synthesized HG material was used in post-combustion CO<sub>2</sub> adsorption systems for carbon capture application. Table 6 presents the synthesis parameters adopted by different groups using HNO<sub>3</sub>.

Xu *et al.* used H<sub>2</sub>O<sub>2</sub> as an etching agent and mixed it with GO. The solution was then heated at 100 °C for 4 h under stirring to obtain holy-GO.<sup>87</sup> The residual H<sub>2</sub>O<sub>2</sub> was later removed by centrifugation and washing of the reaction mixture. The process is shown schematically in Fig. 14. During the heating process, the defective carbon atoms were oxidized and subsequently etched. As a result, pores are formed. The pore size distribution on the basal plane in the range of 2–3 nm was obtained.



Table 5 Nanoparticle-assisted oxidation and pore generation

Nano particle	Starting material	Pre-treatment	Synthesis process	Gas atmosphere	Temp. (°C); time	Final material	Pore size (nm)	SSA (m <sup>2</sup> g <sup>-1</sup> )	Application	Ref.
Ag	Graphene	—	Ag nanoparticles were deposited on the graphene sheets, and then the Ag-coated samples were thermally annealed in air.	Air	300	HG	~5 to tens of nm	590	—	Lin <i>et al.</i> <sup>77</sup>
	GO	GO mixed with AgNO <sub>3</sub> , heat and microwave treatment	Ag-rGO was heat treated, refluxed with nitric acid, and microwave treated.	Ar	300; 2 h	HRGO	2–5	457	LIB	Alsharaeh <i>et al.</i> <sup>78</sup>
Fe <sup>III</sup> porphyrin	HOPG	—	Metalated COF was grown on graphite, oxidized (H <sub>2</sub> O <sub>2</sub> /NaOCl), and COF was removed	—	—	HG	4–50	—	—	Seo <i>et al.</i> <sup>79</sup>
ZnO	GO	—	Layer by layer motif of GO-ZnO was prepared, filtered, and calcined.	Nitrogen	900; 4 h	HG nanosheet	<10	484	Supercapacitor	Wang <i>et al.</i> <sup>80</sup>
Fe <sub>2</sub> O <sub>3</sub>	Graphene	Hydrothermal reaction (180 °C for 12 h)	Calcined in a tube and etched with HCL solution	Argon	850; 2 h	Graphene mesh with pores	60–200	381	LIB	Du <i>et al.</i> <sup>81</sup>
Cu	Graphene	Copper oxide-graphene composite is obtained by the hydrothermal method	The solution was annealed and dilute nitric acid was used to remove the Cu particles	Hydrogen	900; 15 min	HG	Meso: 2.5, 7.6, 11.2, 27.2, 34.7; macro: 52.9; micro: <2	107.8	Supercapacitor	Dutta <i>et al.</i> <sup>82</sup>

Superior capacitive energy storage properties were observed, which were due to the presence of hierarchical 3D porosity in the material.<sup>87</sup> Similarly, the group also produced holey graphene framework (HGF) for electrochemical energy storage applications.<sup>88</sup>

Using a similar process, nanohole GO (NHGO) was synthesized from a homogeneous aqueous mixture of GO and H<sub>2</sub>O<sub>2</sub>.<sup>89</sup>



Fig. 12 Synthesis of HGO and rHGO from GO using HNO<sub>3</sub>. Reprinted with permission.<sup>83</sup> Copyright 2011, American Chemical Society.

The mixture was sonicated for 30 min to prepare the homogeneous solution. The solution was then kept inside an autoclave and hydrothermally treated at 120 °C for 10 h. NHGO with pore size in the range of 2–3 nm was obtained. Other groups carried out similar studies to obtain the porous form of the graphene derivative.<sup>90</sup>

The oxidation process assisted by H<sub>2</sub>O<sub>2</sub> was also reported to create porous graphene from graphene.<sup>91</sup> The oxidation of graphene was initiated by H<sub>2</sub>O<sub>2</sub> and the process could also yield graphene quantum dots. In a separate study, the composite of GO-TiO<sub>2</sub> was prepared by Xu *et al.* using sonication and further heat treated in H<sub>2</sub>O<sub>2</sub> solution to prepare the HG-TiO<sub>2</sub> material for photocatalytic applications.<sup>92</sup>

Wang *et al.* used a microwave reactor to produce HGO.<sup>93</sup> Initially, GO aqueous solution was prepared from graphite flakes using an improved Hummers' method.<sup>94</sup> The GO solution was then mixed with H<sub>2</sub>O<sub>2</sub>, kept in a reaction tube, and irradiated in a microwave reactor at a constant power of 50 W for 45–180 s. The solution was cooled and residual H<sub>2</sub>O<sub>2</sub> was removed *via* centrifugation. The precipitate was re-dispersed in deionized water to form HGO. HGO was further microwave irradiated



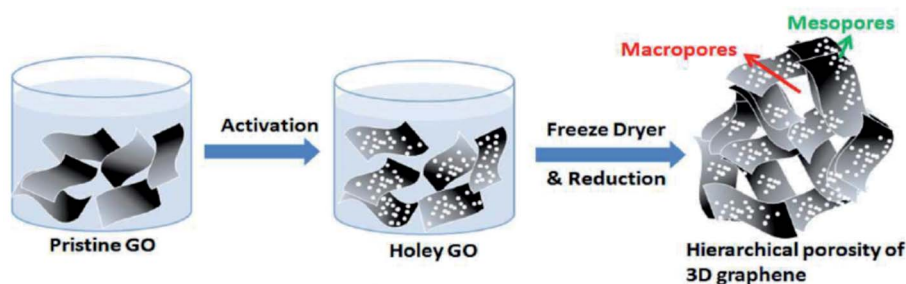


Fig. 13 A three-step approach to obtain porous 3D graphene as demonstrated by Chang *et al.* Reprinted with permission.<sup>85</sup> Copyright 2016, Royal Society of Chemistry.

Table 6 Pore generation using chemical etching process: HNO<sub>3</sub>

Starting Etchant material	Pre-treatment	Synthesis process	Temp. (°C)	Final material	Pore size (nm)	SSA (m <sup>2</sup> g <sup>-1</sup> )	Application	Ref.
HNO <sub>3</sub>	GO	Mixed solution of HNO <sub>3</sub> and GO is sonicated at RT for 1 h	RT	Holey GO	7–600	15	Energy storage	Zhao <i>et al.</i> <sup>83</sup>
	rGO	—	100	Graphene nanomesh	0–100	1010 ± 60	—	Wang <i>et al.</i> <sup>84</sup>
	GO	Mixed solution of GO and HNO <sub>3</sub> (70%) are sonicated for 1 h	—	3D porous graphene	Micropore ~20 nm; nanopore ~2.5 nm	564	Supercapacitor	Chang <i>et al.</i> <sup>85</sup>
	GO	—	—	Holey GO	3–4	524	Carbon capture	Chowdhury <i>et al.</i> <sup>86</sup>

at 220 °C for 90 s to synthesize rHGO. The size of the holes could be controlled with the microwave irradiation time. The rapid synthesis process could be attributed to the *in situ* heat generation process in the sample (GO sheets) during the microwave irradiation. The increased surface area and ion diffusion in the HGO boosts the electrochemical catalytic activity and displays excellent capacitive performance.<sup>93</sup> HG synthesis using H<sub>2</sub>O<sub>2</sub> etching agent by different groups is summarized in Table 7.

Similar to H<sub>2</sub>O<sub>2</sub>, various groups used KOH as an activation agent to create porous graphene/GO material. KOH could react with carbon according to the thermodynamically possible reaction shown below.<sup>95</sup>



The as-produced K and K<sub>2</sub>CO<sub>3</sub> can further react with carbon and create pores. The activation process can generate a continuous 3D network of pores in the range of 1–10 nm.<sup>96</sup> It was found that highly graphitic carbons are not suitable for the KOH activation process. However, the oxygen-functional groups (such as C=O and COOH) attached on the surface can play a crucial role in porosity development. These surface functional species can react with KOH to form an intermediate product of

the C–O–K species. This can further react with the carbon of the precursor material to give a material with a porous structure.<sup>97</sup>

Similarly, Xu *et al.* prepared HG from graphene powder using KOH.<sup>98</sup> In this process, graphene powder was mixed with KOH powder and heated at 800 °C for 1 h in Ar environment to activate the chemical reaction. The experiment resulted in the formation of graphene powder with pores in the range of 1–5 nm. Xia *et al.* prepared 2D porous graphene from exfoliated GO (EGO).<sup>99</sup> The EGO samples were obtained by the thermal

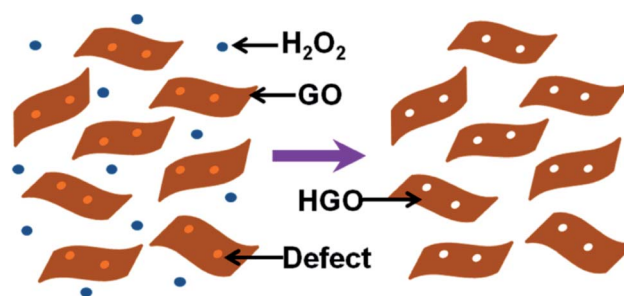


Fig. 14 A schematic diagram to describe the synthesis process of HGO. Reprinted with permission.<sup>87</sup> Copyright 2015, American Chemical Society.



Table 7 Pore generation using the chemical etching process: H<sub>2</sub>O<sub>2</sub>

Etchant	Starting material	Pre-treatment	Synthesis process	Temp. (°C); time	Final material	Pore size (nm)	SSA (m <sup>2</sup> g <sup>-1</sup> )	Application	Ref.
H <sub>2</sub> O <sub>2</sub>	GO	40 mL GO suspension was mixed with 0.4 mL H <sub>2</sub> O <sub>2</sub> and sonicated for 30 min	Hydrothermally treated at 120 °C for 10 h in an autoclave	120	HGO	2–3	425	Supercapacitor	Jinlong <i>et al.</i> <sup>89</sup>
	GO	—	Aqueous mixture is stirred for 4 h	100	HGO	2–3	1330	Supercapacitor	Xu <i>et al.</i> <sup>87</sup>
	Gr	100 mg of graphene was dispersed (sonicated) in 100 mL of H <sub>2</sub> O <sub>2</sub> (30%) for 10 min	The reaction mixture's temperature was increased to 70 °C and kept for 72 h.	70	pGr, GQD	0.7–9	204	Electrocatalyst (oxygen reduction)	Palaniselvam <i>et al.</i> <sup>91</sup>
	GO	GO suspension (20 mL, 2 mg mL <sup>-1</sup> ) mixed with H <sub>2</sub> O <sub>2</sub>	The solution was cured in an autoclave for 6 h	180	Holy graphene hydrogel (HGH)	3D hierarchical porous structure (tens of nm – a few micron)	1261.7	Water desalination	Kong <i>et al.</i> <sup>90</sup>
	GO	—	Mixed solution was stirred at 100 °C for 1, 2, 4, and 4.5 h	100	HGO	34 ± 7 to 62 ± 19	—	Water desalination	Chen <i>et al.</i> <sup>38</sup>
	GO	GO solution was prepared from graphite flakes	Mixed solution of GO and H <sub>2</sub> O <sub>2</sub> was irradiated in a microwave reactor	45–180 s	HGO	13.4 (avg)	—	—	Wang <i>et al.</i> <sup>93</sup>
	GO	GO dispersion was added in TiO <sub>2</sub> suspension	The HGO was further microwave irradiated	220, 90 s	rHGO	—	—	—	—
	GO	GO dispersion was added in TiO <sub>2</sub> suspension	H <sub>2</sub> O <sub>2</sub> solution was added with GO–TiO <sub>2</sub> and heated	160, 4 h	HG–TiO <sub>2</sub> hybrid	—	—	Photocatalysis	Xu <i>et al.</i> <sup>92</sup>

treatment of dried GO at 800 °C. The EGO material was then activated using KOH at 800 °C for 1 h under N<sub>2</sub> flow. The research group also implemented CO<sub>2</sub> to obtain porous graphene from the EGO samples. The CO<sub>2</sub>-activated graphene was reported to have a 3D curly morphology with hierarchical micro-meso-macroscopic structure.

Zheng *et al.* prepared rGO by reducing GO powders prepared by modified Brodie's method.<sup>100</sup> Later, porous activated graphene was synthesized by mixing rGO with KOH at a weight ratio of 1 : 8 and kept at 850 °C for 2 h in Ar atmosphere. The sample was then cleaned with HCl solution, followed by DI water, and then dried to get the activated graphene (porous) sample. It was found that the crystallinity and oxygen functional groups play a crucial role in the growth of the porosity in the activated graphene.

Baburin *et al.* produced perforated graphene from rGO using methanol KOH solution for hydrogen storage application.<sup>101</sup> In an exciting study carried out by Wu *et al.*, a 3D heterostructure architecture combined with holey graphene sheets intercalated with the carbon sphere was fabricated (shown in Fig. 15).<sup>102</sup> rGO and carbon spheres were separately synthesized and mixed through a self-assembly process. Later, the solution was activated at 800 °C for 1 h in Ar atmosphere to create pores on the graphene layers. Due to the unique pillar structure, improved electrical conductivity, higher ionic movement, and diffusion were observed; as a result, the paper electrode made of the

synthesized material showed high energy density and excellent cycling stability for energy storage devices.

Jiang *et al.* used a combined process of chemical etching using KOH and ball milling to produce randomly stacked HG (RSHG).<sup>103</sup> Initially, GO (synthesized using modified Hummers' method) was dispersed in DI water and ultrasonicated for 1 h. The solution was then placed in an autoclave and heated to 180 °C for 12 h to produce the graphene hydrogel. The graphene hydrogel was mixed with KOH and stirred (etched) for 12 h and then ball milled for 30 h. The material was then washed with HCl solution and DI water, and then the RSHG was obtained. The prepared RSHG was demonstrated as a potential anode material for LIB. Pore generation and synthesis of HG using KOH are summarized in Table 8.

## 2.5 Chemical: template-assisted patterning

Periodic holes can be formed on graphene or GO sheets by patterning periodically on the sheets. For example, Zhang *et al.* proposed a method to synthesize nitrogen-doped HG nanocapsules (NHGNs) from GO nanosheets using a template etching process.<sup>104</sup> Amino-functionalized SiO<sub>2</sub> and GO were first assembled using electrostatic interactions (Fig. 16). The process resulted in the formation of GO-encapsulated SiO<sub>2</sub> nanospheres (NSs). The material was then activated in the atmosphere of NH<sub>3</sub>/Ar gas at 900 °C for 90 min (heating rate 10 °C min<sup>-1</sup>).



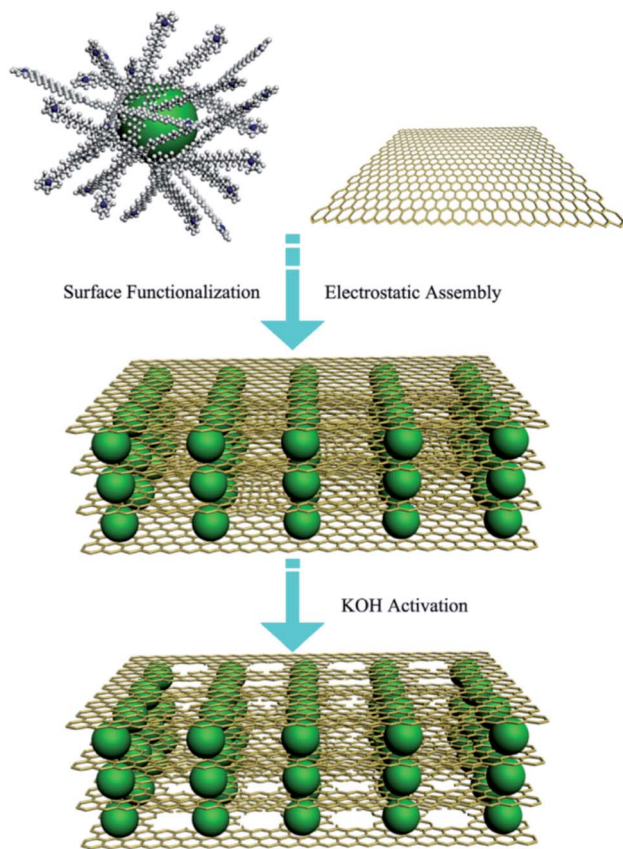


Fig. 15 Schematic picture showing the formation of carbon sphere intercalated holey graphene sheets in a 3D heterostructure architecture form. Reprinted with permission.<sup>102</sup> Copyright 2017, Elsevier.

Subsequently, the obtained material was washed in HF aqueous solution and freeze-dried to get NHGNs.

Ning *et al.* employed a template chemical vapor deposition (CVD) synthesis method to obtain nanomesh graphene (NMG) having nanopores of the size of <math><10\text{ nm}</math>. <sup>105,106</sup> Boiling treatment was first applied to the MgO nanoparticles (20–100 nm) to obtain the Mg(OH)<sub>2</sub> layers. These intermediate layers decompose to give porous MgO layers as a result of water release during calcination. In the next step, methane cracking (CVD) was carried out at 900 °C on porous MgO that finally gives the NMG. This NMG shows extraordinary ferromagnetic properties, which is believed to be due to the high-density defects present in the material. <sup>106</sup> The NMG produced in this way was found to give excellent electrochemical capacitance, rate performance, and cycle stability. <sup>105</sup> The high supercapacitor performance was attributed to the porous nature of the material and the high surface area.

Sinitiskii *et al.* demonstrated a way to create graphene nanomesh by implementing Reactive Ion Etching (RIE) process. <sup>107</sup> As indicated in the schematic Fig. 17, graphene was deposited on copper foils and transferred to an SiO<sub>2</sub>/Si substrate. (a) SiO<sub>2</sub> (10 nm, Mask 1) was deposited and colloidal spheres were prepared using the self-assembly process. (b) RIE 1 (CF<sub>4</sub>) was then used to create gaps of ~10 nm. (c) Au (10 nm, mask 2) was deposited and the colloidal spheres were removed

by ultrasonicing in water. (d) RIE 2 (CF<sub>4</sub> + O<sub>2</sub>) was subsequently used to etch away the exposed graphene regions. (e) Later, mask 1 and mask 2 could be removed by etching in dilute HF. (f) The size of the gaps and the neck widths could be controlled by the RIE parameters.

A similar approach was employed by Wang *et al.* to create graphene nanomesh. <sup>108</sup> Initially, the SiO<sub>2</sub> film was grown on the Cu foil, followed by the formation of polystyrene (PS) nanospheres through a self-assembly process on top of the SiO<sub>2</sub> film. O<sub>2</sub> plasma was used to etch and form gaps in between the closely packed nanospheres. Further, CF<sub>4</sub> plasma was used to remove the exposed SiO<sub>2</sub> regions around the nanospheres. The nanospheres were then dissolved with toluene and removed from the sample. Later, a low-pressure CVD process was used to grow graphene on the exposed Cu foil regions and graphene film with circular pores could be obtained in this way.

Shi *et al.* used spherical porous layered double hydroxide (LDH) and conformally calcined it to get layered double oxide (LDO) microspheres. <sup>109</sup> These LDO microspheres were then used as a template to grow graphene using a high-temperature CVD process, as indicated in the schematic Fig. 18. Later, after a facile chemical etching process, purified graphene micromesh with hierarchical pores was obtained. Brunauer–Emmett–Teller (BET) experiments show a high surface area of 1275 m<sup>2</sup> g<sup>-1</sup>. Hierarchical porous CaO particles were also used as a template to grow hierarchical porous graphene by the research group in a similar fashion. <sup>110</sup> Both the graphene microspheres and the hierarchical porous graphene material were proposed as favorable scaffolds for rechargeable lithium–sulfur battery. Different template-assisted synthesis methods of HG are summarized in Table 9.

### 3. Summary of the synthesis methods

As discussed in the previous sections, different synthesis methods have been proposed and demonstrated to create pores in graphene and graphene oxide in the past few years. These methods or approaches have certain advantages and disadvantages, which impact their potential applications.

Physical methods such as electron or ion beam can be used to create site-specific pores with precise pore size. The graphene with fabricated pores shows promise for DNA sequencing or permeable membranes. Physical methods, however, have a limitation in terms of the mass-scale fabrication of nanopores. On the other hand, thermal annealing methods such as RTA can create porous graphene from GO or graphene powder on a large scale in a short period. Although the GTA process takes time compared to the RTA method, both annealing methods can be used to obtain high-quality HG material that can be further used in different applications. Chemical methods such as chemical etching agent-assisted processes implement etching acids such as H<sub>2</sub>O<sub>2</sub>, HNO<sub>3</sub>, and KOH to create pores in the range from nm to few microns. This method can be used for the mass-scale production of holey/porous graphene materials and the synthesized material can be further applied as a high performance electrode in LIB or supercapacitor. The nanoparticle-assisted (catalytic oxidation) process is another way to



Table 8 Pore generation using the chemical etching process: KOH

Etchant	Starting material	Pre-treatment	Synthesis process	Temp. (°C)	Final material	Pore size (nm)	SSA (m <sup>2</sup> g <sup>-1</sup> )	Application	Ref.
KOH	GO	Microwave exfoliated GO (MEGO) and thermally exfoliated GO are prepared	MEGO was chemically activated by mixing with KOH	—	Activated MEGO	1–10	3100	Supercapacitor	Zhu <i>et al.</i> <sup>96</sup>
	Graphite oxide	Graphene powder was produced from graphite oxide by a vacuum-promoted low-temp. Exfoliation	The graphene powder was mixed with KOH and heated for 1 h under Ar atmosphere	800	HG powder	1–5	2300	Conductive additive in the LiFePO <sub>4</sub> cathode	Xu <i>et al.</i> <sup>98</sup>
	GO	Exfoliated GO is prepared from GO using thermal treatment	Activation was done at 800 °C for 1 h under N <sub>2</sub> flow	800	HG (2D lamellas)	Bimodal micro-mesopore	2518	Supercapacitor	Xia <i>et al.</i> <sup>99</sup>
	rGO	GO powders were reduced at 600 °C for 30 s in Ar atmosphere	rGO mixed with KOH (1 : 8) and chemically activated at 850 °C for 2 h in Ar atmosphere, cleaned, and dried	850	Porous activated graphene	—	2406	Supercapacitor	Zheng <i>et al.</i> <sup>100</sup>
	rGO	Thermal exfoliation of GO is done to get rGO	rGO powder mixed with methanol KOH solution, dried and annealed at 800 °C for 1 h	800	Perforated graphene	~1.7	2900	Hydrogen storage	Baburin <i>et al.</i> <sup>101</sup>
	rGO, carbon sphere	rGO and carbon sphere (CS) are prepared separately	rGO and CS are mixed through a self-assembly process and activated using KOH for 1 h	800	HG intercalated with CS	2–10	288.19	Electrochemical capacitor	Wu <i>et al.</i> <sup>102</sup>
	GO	Graphene hydrogel was produced by mixing GO and DI water and heating	The graphene hydrogel was mixed with KOH, stirred, and ball milled	—	Randomly stacked HG	—	800	LIB	Jiang <i>et al.</i> <sup>103</sup>

synthesize HG from materials such as GO or graphene. This method uses nanoparticles as a catalyst to synthesize HG. Various groups have also used templates to make patterned holes/pores on graphene sheets. However, template preparation and removal can be tedious and may not be suitable for mass-scale HG fabrication. Table 10 summarizes the different synthesis methods discussed above, the major parameters to control the pore size distribution, the advantages and limitations of the methods, and the major fields where the synthesized materials are used.

The bar chart in Fig. 19 further summarizes the scale of the pore size created using different synthesis methods. The high-intensity color value in a bar indicates a higher number of data points or works reported in the literature in a particular pore size range. For example, in the case of chemical methods, the high intensity lies within 10 nm and few works reported pore generation of few hundreds of nm. Similarly, the thermal annealing process mainly allows the generation of pores of few nm to few hundreds of nm. If GTA is used, it, in general, can produce pores of few nm to few tens of nm and the RTA method could generate pores of few nm to few hundreds of nm. Likewise, other bars are also presented and shown in the bar chart. Although different methods can produce pores of various sizes, it can be noticed from the bar chart that the most common or

achievable pore sizes are in the range of few nm to tens of nm. The smallest possible pore size that could be created is a few Angstroms (<1 nm),<sup>53,56,91</sup> which could be obtained using the electron/ion beam or chemical etching (H<sub>2</sub>O<sub>2</sub>) process. On the other hand, the biggest pore size can be of few tens of micron.<sup>76</sup>

### 3.1 Controlling the pore size distribution in HG

Having precise control over the pore size distribution, particularly the pore diameter, shape, and periodicity, is essential for the reliable and quality synthesis of HG. When E-beam is used to create micro/nanopores, the pore size can be directly controlled by controlling the beam diameter used in the SEM or TEM. Generally, the intensity in an E-beam follows a Gaussian shape and usually, the size of the beam or, more precisely, the full-width half maximum (FWHM) can be controlled in the microscope itself through the lens optics. In addition, the beam diameter also depends on the beam energy. Thus, the beam energy could play a significant role if a precise hole is needed. Similarly, in the ion beam-assisted pore fabrication process, the pore size can be determined with the beam diameter. The beam energy and the ion type used could also impact the pore size.<sup>59,60</sup>



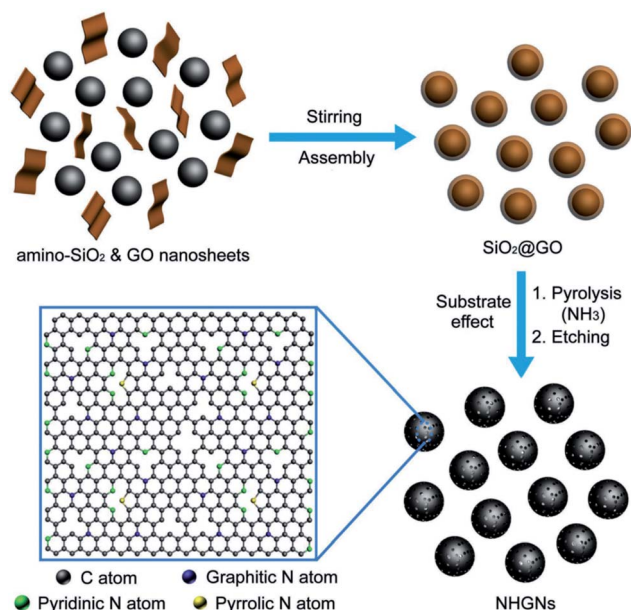


Fig. 16 Schematic figure showing the steps for the synthesis of NHGNs. Reprinted with permission.<sup>104</sup> Copyright 2016, John Wiley and Sons.

The annealing method can induce a large number of pores in a short time. The pore size distribution of the grown pores is highly dependent on the heating rate and the temperature. It is found that the size of the pores, as well as the hole density, tend to increase with the heating/ramping rate and the temperature.<sup>63,64,69</sup> In addition, the holding time could also impact the pore size distribution.<sup>71</sup>

When chemical etching agents such as, KOH, HNO<sub>3</sub>, or H<sub>2</sub>O<sub>2</sub> are used to fabricate pores, the pore size distribution depends on the reaction time. Usually, increasing the reaction time would increase the pore size in the material.<sup>84,87</sup> The pore size

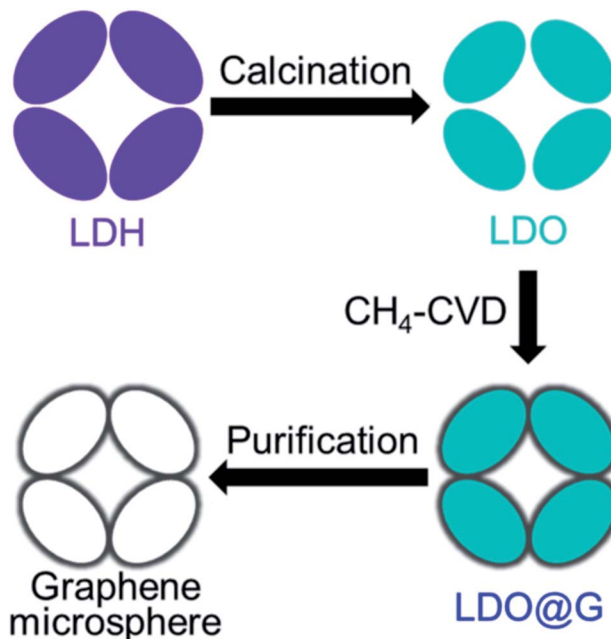


Fig. 18 Schematic picture indicating the steps to grow graphene microspheres. Reprinted with permission.<sup>109</sup> Copyright 2015, Elsevier.

distribution in the nanoparticle-assisted fabrication process follows the nanoparticle size distribution used during the synthesis process. Thus, the pore size in the material can be tailored with the size of the nanoparticles. On the other hand, the template-assisted chemical etching approach can create well-defined pores with specific periodicity. Consequently, the pore size and pore density can be precisely controlled when the template-based etching process is implemented.

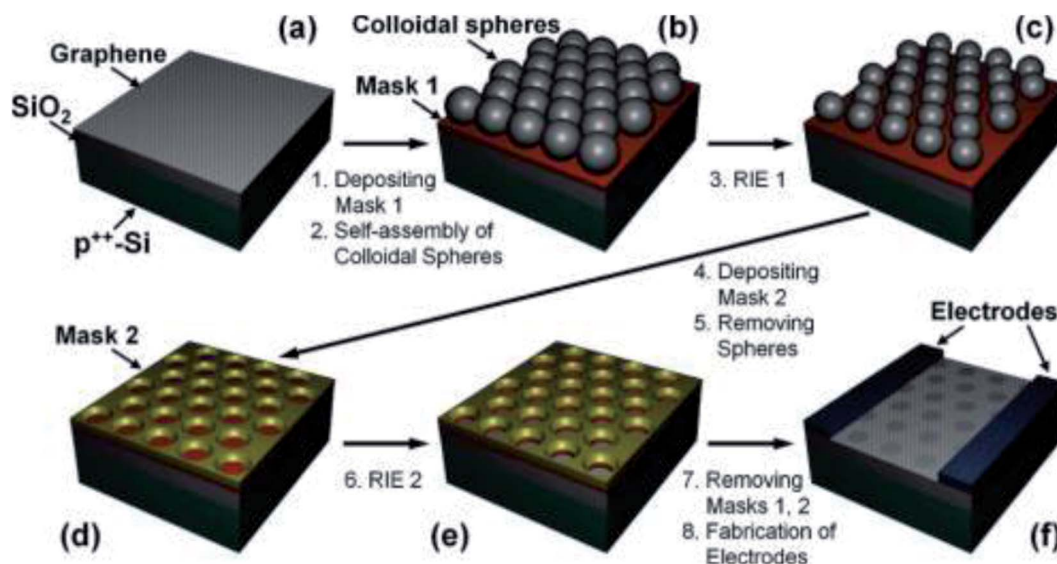


Fig. 17 Schematic of the nanomesh fabrication process using colloidal microspheres and RIE. Reprinted with permission.<sup>107</sup> Copyright 2010, American Chemical Society.





Table 9 Pore generation using the template-assisted process

Template used	Starting material	Post growth	Final material	Pore size (nm)	SSA ( $\text{m}^2 \text{g}^{-1}$ )	Applications	Ref.
Amino-functionalized $\text{SiO}_2$	GO	—	N-Doped HG nanocapsules	—	1162	Electrocatalysis	Zhang <i>et al.</i> <sup>104</sup>
Porous MgO layers	—	Graphene growth ( $\text{CH}_4$ cracking, CVD)	Nanomesh graphene	<10	1600–2000	Ferromagnetism	Ning <i>et al.</i> <sup>106</sup>
Porous MgO layers	—	Graphene growth ( $\text{CH}_4$ cracking, CVD)	Nanomesh graphene	<10	1654	Supercapacitor	Ning <i>et al.</i> <sup>105</sup>
Colloidal microspheres	Graphene	—	Graphene nanomesh	~100–few hundred	—	—	Sinitskii <i>et al.</i> <sup>107</sup>
Polystyrene nanospheres	—	Graphene growth (LPCVD)	Graphene nanomesh	40	—	Transistor	Wang <i>et al.</i> <sup>108</sup>
Layered double oxide microspheres	—	Graphene growth (CVD of $\text{CH}_4$ )	Porous graphene microsphere	Hierarchical pores	1275	Lithium–sulfur battery	Shi <i>et al.</i> <sup>109</sup>
Hierarchical porous CaO particles	—	Graphene growth (CVD of $\text{CH}_4$ )	Porous graphene	Hierarchical pores	572	Lithium–sulfur battery	Tang <i>et al.</i> <sup>110</sup>

### 3.2 Controlling defects in HG

Synthesizing HG with controlled defects is an important aspect. Creating vacancies by knocking carbon atoms using an electron beam could result in various processes; for example, a nearby carbon atom can fill up the vacancy or diffusion of the vacancies could take place. Because they exhibit temperature-dependent self-repairing processes, the defects can be controlled by regulating the substrate temperature.<sup>49</sup> The study also suggests that the stability of pores has complex dynamics and that zigzag-type edge defects tend to be more stable than armchair defects.<sup>51</sup> Molecular dynamics study predicts that when the ion beam is used for pore creation, the beam type and energy could impact the defect formation process in graphene.<sup>60</sup> In the case of high-energy ions irradiation, increasing the ion energy reduces the number of structural defects formed in the few-layer graphene films.<sup>59</sup> When the annealing process is used for the generation of pores/holes, it is found that the ramping rate and the annealing temperature can impact the defect density in HG. The defect density in HG nanosheets increases with the increase in the ramp rate and the annealing temperature.<sup>64,69</sup>

During chemical etching, for example, when  $\text{H}_2\text{O}_2$  is used to create pores from GO sheets, the defects/disorders are found to increase with the doses of  $\text{H}_2\text{O}_2$ .<sup>92</sup> Similar results are also observed when GO is treated with KOH solution.<sup>99</sup> The study indicates that more defects are found on the graphene plane when the KOH/GO mass ratio is increased. During the nanoparticle-assisted etching process such as etching GO with ZnO to synthesize holey graphite nanosheets, a higher amount of the catalyst (ZnO) can increase the number of structural defects in the synthesized material.<sup>80</sup> In numerous applications, such as in batteries, such defects could help to enhance the electrochemical performance of the cell.<sup>81</sup> On the other hand,

when nanoparticles or micro/nanospheres are used as a template to create tailored pores, the self-assembly of the particles induces various defects in the material,<sup>107</sup> prompting further studies to control the defect formation process.

### 3.3 Other promising synthesis routes to fabricate HG

In addition to the above discussed commonly reported methods for creating or synthesizing pores in graphene/GO materials, there are also other methods that are being experimented by different research groups. The green synthesis method is a promising approach for the synthesis of HG/porous graphene where various biomaterials such as, Bougainvillea flowers, fungus, biomass or leaves of *Plumeria rubra* have been used as the starting raw material.<sup>111–113</sup> This method is environmentally friendly and likely to be cost-effective as compared to other



Fig. 19 The bar chart shows the different synthesis methods used to create pores of different sizes.



Table 10 A comparison among the different synthesis methods of HG

Type	Synthesis method	Pore size	Controlling the pore size distribution	Advantages	Limitations/Disadvantages	Major areas of application
Thermal	Thermal annealing	Few nm–few hundreds of nm	Heating rate, temperature, holding time	Fast (RTA), mass-scale production	GTA usually takes hours	Energy storage (LIB, supercapacitor)
Physical	Electron & ion beam	<10 nm–1 $\mu\text{m}$	Beam diameter, beam energy ion beam type	Precise pore size	Small area fabrication	Atomic-scale application (permeable membrane, DNA sequencing)
Chemical	Chemical etching	Few nm–few microns	Reaction time	Mass scale production is possible	The produced material could have impurities	Energy storage (LIB, supercapacitor), water desalination
	Nanoparticle-assisted oxidation	Few nm–few hundreds of nm	Nanoparticle size	The pore size could be tailored with the size of the nanoparticles	Could be inefficient in mass production	Energy storage (LIB, supercapacitor)
	Template directed patterning	Regular pore: <10 nm; hierarchical pores	Etching parameters	Pore size and pore density can be controlled	Template preparation and removal could be complicated	Energy storage

methods. Electrochemical exfoliation<sup>114</sup> and nanoimprint lithography<sup>115</sup> methods are also used for the synthesis of HG. Other reported methods such as the sol-gel technique have been used to prepare porous graphene aerogel for capacitor application<sup>116</sup> and laser engraving was implemented to create 3D hierarchically porous graphene for supercapacitor electrode application.<sup>117</sup> The combustion method is another approach employed to fabricate porous graphene from GO.<sup>118</sup> A bottom-up strategy was devised by Moreno *et al.* to create nanoporous graphene, which showed semiconducting and nanosieving functionalities.<sup>119</sup> The trend in the synthesis of the HG material using different approaches as observed in the literature indicates that HG can be obtained by various ways and many of these methods are relatively cheaper with specific advantages and drawbacks.

### 3.4 Tools and techniques to characterize HG

Various techniques have been used for the appropriate characterization of HG. Raman spectroscopy has been one of the primary effective techniques for the structural and chemical characterization of HG. It can essentially figure out the different Stokes' band of graphitic, non-graphitic, and defect-related peaks in such materials. The peak appearing at  $\sim 1350\text{ cm}^{-1}$  in the Raman spectrum is corresponding to the D-band disorder (defects) in carbon material and the peak that usually appears at  $\sim 1580\text{ cm}^{-1}$  is the G-band corresponding to the in-plane stretching of  $\text{sp}^2$  carbon atoms, which signifies the graphitic nature of the carbon material. Another major peak can be often noticed to appear at  $\sim 2700\text{ cm}^{-1}$ , which is a secondary D peak and is assigned as the 2D band, mostly found pronounced in case of monolayer or few layers of graphene; in case of vertically-oriented multilayer graphene sheets, the intensity of the peaks will be determined by the terminated layers.<sup>120</sup> Usually, the relative intensity of the D to G peaks can be used to understand the graphitic property of a material. For example, in the synthesized HG material (Fig. 6), the relative intensity of D to G

Raman peaks were used to conclude the removal of the defective carbons and formation of quality (highly graphitic) HG material. Thus, by comparing the intensity, their relative positions, and the FWHM of the peaks, various structural properties of the synthesized HG can be evaluated, as demonstrated by various works.<sup>59,62,63,93,121</sup>

Apart from Raman spectroscopy, electron microscopy tools such as SEM and TEM can be used for the assessment of the morphology or to observe the porous nature of the HG materials. On the one hand, SEM can be used for a quick understanding of the material, while TEM can offer intricate information related to the shape, size, or the structural nature of the formed pores in the material.<sup>48,77,82</sup> TEM-based electron energy loss spectroscopy (EELS) method can provide chemical information related to the  $\pi$  and  $\sigma$  bonds in the material and their contribution to the material; thus, the presence of  $\text{sp}^2$  and  $\text{sp}^3$  carbons can be evaluated with atomic scale spatial resolution.<sup>96</sup> Techniques such as X-ray diffraction (XRD) could be used to understand the crystalline nature of the material, variation of the interlayer spacings of HG with the synthesis parameters, *etc.*<sup>73,91,100</sup> Another important parameter that is often required is the evolution of the porosity or distribution of the pores in the material. BET or the Barret–Joyner–Halenda (BET/BJH) method is mostly considered for the evaluation of such information from an HG material. Information such as pore size distribution, average pore size, pore density, *etc.*, can be obtained from BET/BJH analysis.<sup>11,71,83</sup>

## 4. Applications of HG

Due to the various exciting and unique properties of HG, they have been considered in different fields, mainly focusing on energy storage, water permeance and desalination, carbon capture, and DNA sequencing.

The high surface area of HG/porous graphene and the presence of interchannels for ionic/molecular movement are



unique properties of HG. In addition, the interlayer attractions in HG are relatively weaker due to the lack of sufficient number of carbon atoms and the interlayer distance is higher than other forms of graphene. These properties instigated its use instead of graphene in lithium-ion batteries (LIB). Conventional graphite-based LIBs have a theoretical specific capacity of  $372 \text{ mA h g}^{-1}$ . This value can be improved by changing the anode materials such as carbon nanotubes, porous carbons, or graphene. Graphene can provide higher capacity due to its high specific surface area. A single layer of graphene has a theoretical surface area of  $2630 \text{ m}^2 \text{ g}^{-1}$ . However, the performance of the graphene anode is hindered as the graphene layers tend to restack, which results in the decrease in the available surface area for Li-ion intercalation. Hence, HG, when used as an anode/electrode in batteries or capacitors, significantly enhances the efficiency of the device, as demonstrated by various works.<sup>41,42,74,87,88,122–125</sup> It is now well known that the degree of microporosity, specific surface area, and surface wettability of a grown porous graphene material play a key role in the capacitance in a capacitor.<sup>99</sup> The efficiency of HG as an energy storage electrode can be further enhanced by doping with other elements such as nitrogen, sulfur, and phosphorus.<sup>65,126,127</sup>

DNA sequencing is currently an important field in biological sciences, which can benefit from the development of nanopore graphene membranes.<sup>128</sup> When DNA passes through a membrane, there are changes in the ionic current. This change depends on the shape, size, and length of the DNA.<sup>129–131</sup> The ideal size of a nanogap for DNA to pass through is a few nm,  $\sim 1.0\text{--}1.5 \text{ nm}$ ,<sup>132</sup> which could be achieved in the case of HG/porous graphene. The most effective method to create precise

nanopores for DNA sequencing has been found to be the electron or ion beam-assisted physical removal of carbon atoms.<sup>52,61,133,134</sup> However, the created nanopores could be unstable due to the presence of unsaturated bonds. It has been reported that by passivating the dangling carbon bonds with Si atoms, these pores could be made stable.<sup>135</sup>

Another primary application of HG or porous graphene is in water desalination. The porous nature of the material can be used for water permeance and hence in the production of fresh water from impure water. Several studies indicated the effectiveness of using holey GO and HG in such applications.<sup>15,16,38,136,137</sup> When ion-containing water is allowed to pass through the graphene sheets, the larger ions can be blocked from passing through the pores. In this case, the pore size distribution is the critical factor determining the permeation of water molecules but not the ions. This is shown schematically in Fig. 20a.

Along with the porous nature of the material, the interlayer distance, particularly in GO or rGO, can also play a significant role in the water desalination process.<sup>28,33,138–140</sup> GO/rGO membranes are found to be stable in water<sup>141,142</sup> and have high water permeance properties.<sup>143–145</sup> These properties of GO or rGO can be further enhanced or modulated by creating pores in the materials. This is shown in the schematic Fig. 20b below, where stacked graphene oxide (GO) acts as a filter for water molecules.

Solar power-assisted water purification is another advanced and promising field of research for an eco-friendly and sustainable future. 2D nanomaterials are suitable for such applications due to their efficient photothermal, broad band absorption, and photocatalytic transformation, in addition to



Fig. 20 Water purification via ion filtering using nanoporous graphene (a) and stacked graphene oxide (b). Reprinted with permission.<sup>33</sup> Copyright 2015, Royal Society of Chemistry.



the larger surface area that they can offer.<sup>146,147</sup> A recent study by Yang *et al.* demonstrated a solar energy convertor for water desalination using a 3D cross-linked polymer-like (porous-graphene) material.<sup>148</sup> In a separate work, a membrane of porous graphene and polyimide was prepared with a 3D architecture.<sup>149</sup> The membrane with the 3D structure displayed excellent solar absorption capability; on the other hand, the 3D porous network boosts water vapor escape. Ito *et al.* designed a porous graphene material-based steam generator.<sup>150</sup> The generator can convert sunlight into high-energy steam at a promisingly high energy efficiency of 80%.

It is also reported that photocatalytic activity of a photocatalyst material such as TiO<sub>2</sub> can be further enhanced if the photocatalyst material is mixed with HG.<sup>92</sup> The presence of holes allows light permeation and reactant diffusion, and enhances the photocatalytic properties. While using in combination with ZnO nanoparticles, porous graphene promotes the effective charge separation of electron-hole pairs and increases the photocatalytic performance.<sup>151</sup> Similarly, nanocomposites of three-dimensional porous graphene and Co<sub>3</sub>O<sub>4</sub> were used for high-performance photocatalysis.<sup>152</sup>

The porous form of graphene can also help in the storage of hydrogen.<sup>153</sup> It is believed that the hydrogen storage mechanism varies from sample to sample.<sup>154</sup> The interaction between hydrogen and carbon materials is governed by van der Waals force. This attractive force could lead to physisorption. If hydrogen comes very close to the carbon atoms, then this could lead to the overlapping of wave functions of carbon atoms with those of the hydrogen electron. This can lead to the chemisorption of hydrogen on the carbon material.

Graphene membranes with nanopores have been applied in the field of gas separation for the filtration of CO<sub>2</sub> gas molecules from other gases. This was achieved by tailoring the pores to allow the CO<sub>2</sub> molecules to pass while blocking other gases such as NO<sub>2</sub>.<sup>155,156</sup>

In general, the scale of the pore size could impact on the field of application of the HG. This is visually represented in the bar chart in Fig. 21. The chart is a summary of the different major synthesis methods and tables discussed in this review paper. The intensity in the bars is corresponding to the number of

results reported. The chart provides a rough estimation of the scale of pores created for specific applications. In the supercapacitor application, the pore sizes designed are in the range of few nm and mostly <10 nm. Similarly, the pores created for battery application are in the range of a few nm to few hundreds of nm. For DNA sequencing and translocating, pores with a small size in the range of a few nm are needed, which could be precisely fabricated using an energetic electron or ion beam. For water desalination applications, the typical pore size depends on the type of desalination technology the membrane is intended for; *e.g.*, reverse osmosis membranes require pore sizes less than 1 nm for efficient salt removal, whereas nano or micro-filtration membranes typically have larger pores. Similarly, for application in carbon capture, the scale of the pores is narrower and can be few nm.

In addition to the above-mentioned areas where HG has been successfully used, the unique structural configuration and properties of HG/porous graphene have found various potential applications in areas such as in solar cells,<sup>157,158</sup> field-effect transistors (FET),<sup>115,159</sup> fuel cells,<sup>160</sup> and microwave absorption.<sup>161–164</sup> Graphene is a desirable material for gas sensing applications.<sup>165–167</sup> By creating pores in the graphene or GO material, the sensing performance can be efficiently enhanced.<sup>168,169</sup> There are also other unique applications of HG. For example, a high-efficiency thermocell device was designed by Li *et al.* that was capable of working at temperatures as low as –40 °C.<sup>170</sup> HG was also demonstrated as an efficient planar field emitter.<sup>171</sup> The applied field concentrates along the circumference of the holes and provides field emission sites. In an interesting work by Chen *et al.*, an hrGO-based FET type device was made to detect the bacteria.<sup>172</sup> The hrGO material with p-type semiconductor properties was used as the transducer element in the FET device. The device could detect *E. coli* bacteria with a detection limit as low as 803 cfu mL<sup>–1</sup>. In a separate work, graphene oxide nanomesh (GONM) was synthesized to treat cancer cells.<sup>173</sup> GONM was initially developed using photocatalytic degradation and subsequently functionalized by mixing with polyethylene glycol (PEG), arginine-glycine-aspartic acid (RGD)-based peptide and cyanine 7 (Cy7). The composite has strong absorption in the NIR range and thus, when the tumor cells targeted by the composite were irradiated with low power laser, cell ablation occurred. In this way, the cancer cells could be treated successfully. The unique material properties and versatile nature of applications of HG make it a potential field for researchers to investigate and explore further.

Similar to graphene with pores, other 2D materials with pores could be fabricated and used in different fields. For example, 2D materials such as Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> foam with pores could be used as the electromagnetic shielding material.<sup>174</sup> In the optoelectronics sector, 2D semiconductors have shown promising results, mainly due to their exhibition of many-body complexes such as exciton, trion, and biexciton.<sup>175</sup> Optical properties along with their device applications of several 2D materials are well-reviewed.<sup>176–178</sup> Modulating these properties by engineering the materials could open up new avenues of applications. Similarly, the optoelectronic properties of HG have not been well studied



Fig. 21 Bar chart showing the different fields of application vs. the pore size of HG being tested.



and remain an undiscovered field that needs more attention. MXene is an emerging 2D material and is a promising field for several types of applications.<sup>179</sup> The review paper<sup>179</sup> discusses various exciting properties of MXene along with several other applications of porous MXenes such as porous MXene with hierarchical structures for enhanced oxygen evolution reaction (OER), 3D interconnected porous frameworks of MXene with a high specific area for energy storage, and EMI applications. The studies suggest that similar to the emerging 2D semiconductor field, HG with tunable properties is a potential and untouched field that needs more research.

## 5. Conclusions and future outlook

HG is an emerging new material with exciting and unique properties that could find potential applications in different fields including semiconductor industry, water purification, solar cell, fuel cell, DNA sequencing, and energy storage. Several synthesis methods have been developed over the past few years and in this review work, we have summarized the most promising approaches and their technological applications in different fields. The synthesis/fabrication methods are systematically categorized to benefit the readers and help them understand the type of holes/pores generated, various parameters to control the pore size distribution, and their potential applications in different fields. In specific applications, such as DNA sequencing, devices have to be made out of a monolayer or few layers of graphene, for which the synthesis of high-quality graphene is needed. However, for most of the other applications, the bulk material of holey/porous graphene could be prepared at a significantly low cost. Synthesis methods such as annealing or chemical approaches have the potential to produce HG in the bulk form, which could be later used in the field of energy storage, water purification, *etc.* In these methods, GO or graphitic powder can be used as a starting material. GO has been a common starting material for the synthesis of HG and due to the low production cost of GO; the overall cost of HG synthesis could be a fraction of that of CVD-grown graphene.<sup>180–182</sup> In addition, most synthesis methods exploit the presence of structural defects in the graphene layers. Such defects have been reported to facilitate the formation of pores. Accordingly, the precursor for HG or HGO does not need to be the most expensive form of graphene.

Despite being a promising emerging versatile material, there are certain aspects of HG that the research community should address before it can be commercially produced for various applications. One of the aspects is the systematic study of the pore production to tailor and generate precise pores by cost-effective methods such as annealing since a well-defined pore distribution is highly desirable in many applications. Understanding the structural degradation of the synthesized HG under certain conditions such as long-term performance in water purification or the change in the inherent optoelectronic property with the environment is also essential. Broader in-depth studies focused on specific applications are thus needed. HG and composites of HG have shown promising results in energy storage applications primarily due to the

increased specific surface area. This field can be further developed and thoroughly researched with the combination of porous/HG and nanoparticles with hierarchical sizes (<2 nm to a few  $\mu\text{m}$ ). The relation between the performance and pore size distribution as well as the nanoparticle size needs to be understood. At last, there are more approaches that are yet to be developed that could produce well-defined holey/porous graphene material, which should be economical, less time consuming, and thus, should have mass-scale production capability.

## Data availability

Data will be provided upon request.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 M. Sang, J. Shin, K. Kim and K. J. Yu, *Nanomaterials*, 2019, **9**, 374.
- 2 D. G. Papageorgiou, I. A. Kinloch and R. J. Young, *Prog. Mater. Sci.*, 2017, **90**, 75–127.
- 3 A. K. Geim and K. S. Novoselov, *Nat. Mater.*, 2007, **6**, 183–191.
- 4 L. Lin, H. Peng and Z. Liu, *Nat. Mater.*, 2019, **18**, 520–524.
- 5 C. Dean, A. F. Young, L. Wang, I. Meric, G. H. Lee, K. Watanabe, T. Taniguchi, K. Shepard, P. Kim and J. Hone, *Solid State Commun.*, 2012, **152**, 1275–1282.
- 6 M. Yankowitz, Q. Ma, P. Jarillo-Herrero and B. J. LeRoy, *Nat. Rev. Phys.*, 2019, **1**, 112–125.
- 7 A. K. Geim and I. V. Grigorieva, *Nature*, 2013, **499**, 419–425.
- 8 F. Ye, J. Lee and P. X. L. Feng, *Nanoscale*, 2017, **9**, 18208–18215.
- 9 P. M. Pataniya and C. K. Sumesh, *ACS Appl. Nano Mater.*, 2020, **3**, 6935–6944.
- 10 Y. Tao, Z.-Y. Sui and B.-H. Han, *J. Mater. Chem. A*, 2020, **8**, 6125–6143.
- 11 E. D. Walsh, X. Han, S. D. Lacey, J.-W. Kim, J. W. Connell, L. Hu and Y. Lin, *ACS Appl. Mater. Interfaces*, 2016, **8**, 29478–29485.
- 12 Z. Xing, J. Tian, Q. Liu, A. M. Asiri, P. Jiang and X. Sun, *Nanoscale*, 2014, **6**, 11659–11663.
- 13 S. P. Surwade, S. N. Smirnov, I. V. Vlassiuk, R. R. Uncic, G. M. Veith, S. Dai and S. M. Mahurin, *Nat. Nanotechnol.*, 2015, **10**, 459–464.
- 14 A. Boretti, S. Al-Zubaidy, M. Vaclavikova, M. Al-Abri, S. Castelletto and S. Mikhalovsky, *npj Clean Water*, 2018, **1**, 5.



- 15 Z. Fa, *IOP Conf. Ser.: Mater. Sci. Eng.*, 2020, **926**, 012011.
- 16 D. Cohen-Tanugi and J. C. Grossman, *Nano Lett.*, 2012, **12**, 3602–3608.
- 17 R. O. Brennan, *J. Chem. Phys.*, 1952, **20**, 40–48.
- 18 T. Gould, K. Simpkins and J. F. Dobson, *Phys. Rev. B*, 2008, **77**, 165134.
- 19 G. L. Klimchitskaya, U. Mohideen and V. M. Mostepanenko, *Phys. Rev. A*, 2000, **61**, 062107.
- 20 L. G. MacDowell, *J. Chem. Phys.*, 2019, **150**, 081101.
- 21 G. L. Klimchitskaya and V. M. Mostepanenko, *Phys. Rev. B*, 2013, **87**, 075439.
- 22 Y. Ding, Y. Wang, S. Shi and W. Tang, *J. Phys. Chem. C*, 2011, **115**, 5334–5343.
- 23 A. Du, Z. Zhu and S. C. Smith, *J. Am. Chem. Soc.*, 2010, **132**, 2876–2877.
- 24 D. Singh, V. Shukla and R. Ahuja, *Phys. Rev. B*, 2020, **102**, 075444.
- 25 P. A. Denis, R. Faccio and F. Iribarne, *Comput. Theor. Chem.*, 2012, **995**, 1–7.
- 26 L. Huang, S. Miao, X. Wang and X. Yang, *Mol. Simul.*, 2020, **46**, 853–863.
- 27 P. A. Denis and F. Iribarne, *J. Phys. Chem. C*, 2013, **117**, 19048–19055.
- 28 Y. Qian, X. Zhang, C. Liu, C. Zhou and A. Huang, *Desalination*, 2019, **460**, 56–63.
- 29 T. Liu, L. Tian, N. Graham, B. Yang, W. Yu and K. Sun, *Environ. Sci. Technol.*, 2019, **53**, 11949–11959.
- 30 F. Changjing, G. Zhao, H. Zhang and S. Li, *Int. J. Electrochem. Sci.*, 2013, **8**, 6269–6280.
- 31 S. Rao, J. Upadhyay, K. Polychronopoulou, R. Umer and R. Das, *J. Compos. Sci.*, 2018, **2**, 25.
- 32 A. Abid, P. Sehrawat, S. S. Islam, P. Mishra and S. Ahmad, *Sci. Rep.*, 2018, **8**, 3537.
- 33 F. Perreault, A. Fonseca de Faria and M. Elimelech, *Chem. Soc. Rev.*, 2015, **44**, 5861–5896.
- 34 B. Lesiak, G. Trykowski, J. Tóth, S. Biniak, L. Kövér, N. Rangam, L. Stobinski and A. Malolepszy, *J. Mater. Sci.*, 2021, **56**, 3738–3754.
- 35 V. H. Luan, H. N. Tien, L. T. Hoa, N. T. M. Hien, E.-S. Oh, J. Chung, E. J. Kim, W. M. Choi, B.-S. Kong and S. H. Hur, *J. Mater. Chem. A*, 2013, **1**, 208–211.
- 36 A. M. Selim, M. A. Wasfey, H. H. Abdulllah, M. B. Zahran, W. Khalifa and I. S. El-Mahallawi, *2019 6th International Conference on Advanced Control Circuits and Systems (ACCS) & 2019 5th International Conference on New Paradigms in Electronics & information Technology (PEIT)*, 2019, pp. 94–98.
- 37 K. A. Sammed, L. Pan, M. Asif, M. Usman, T. Cong, F. Amjad and M. A. Imran, *Sci. Rep.*, 2020, **10**, 2315.
- 38 X. Chen, Z. Feng, J. Gohil, C. M. Stafford, N. Dai, L. Huang and H. Lin, *ACS Appl. Mater. Interfaces*, 2020, **12**, 1387–1394.
- 39 M. Yang, Y. Wang, L. Dong, Z. Xu, Y. Liu, N. Hu, E. S.-W. Kong, J. Zhao and C. Peng, *Nanoscale Res. Lett.*, 2019, **14**, 218.
- 40 J. Zhao, Y.-Z. Zhang, F. Zhang, H. Liang, F. Ming, H. N. Alshareef and Z. Gao, *Adv. Energy Mater.*, 2019, **9**, 1803215.
- 41 X. Hu, D. Bai, Y. Wu, S. Chen, Y. Ma, Y. Lu, Y. Chao and Y. Bai, *Chem. Commun.*, 2017, **53**, 13225–13228.
- 42 T. Liu, L. Zhang, B. Cheng, X. Hu and J. Yu, *Cell Rep. Phys. Sci.*, 2020, **1**, 100215.
- 43 A. C. Lokhande, I. A. Qattan, C. D. Lokhande and S. P. Patole, *J. Mater. Chem. A*, 2020, **8**, 918–977.
- 44 Y. Lin, Y. Liao, Z. Chen and J. W. Connell, *Mater. Res. Lett.*, 2017, **5**, 209–234.
- 45 Y. Zhang, Q. Wan and N. Yang, *Small*, 2019, **15**, 1903780.
- 46 P. Ramos Ferrer, A. Mace, S. N. Thomas and J.-W. Jeon, *Nano Convergence*, 2017, **4**, 29.
- 47 J. Rouquerol, D. Avnir, C. W. Fairbridge, D. H. Everett, J. M. Haynes, N. Pernicone, J. D. F. Ramsay, K. S. W. Sing and K. K. Unger, *Pure Appl. Chem.*, 1994, **66**, 1739–1758.
- 48 M. D. Fischbein and M. Drndić, *Appl. Phys. Lett.*, 2008, **93**, 113107.
- 49 B. Song, G. F. Schneider, Q. Xu, G. Pandraud, C. Dekker and H. Zandbergen, *Nano Lett.*, 2011, **11**, 2247–2250.
- 50 D. Fox, A. O'Neill, D. Zhou, M. Boese, J. N. Coleman and H. Z. Zhang, *Appl. Phys. Lett.*, 2011, **98**, 243117.
- 51 Ç. Ö. Girit, J. C. Meyer, R. Erni, M. D. Rossell, C. Kisielowski, L. Yang, C.-H. Park, M. F. Crommie, M. L. Cohen, S. G. Louie and A. Zettl, *Science*, 2009, **323**, 1705.
- 52 G. F. Schneider, S. W. Kowalczyk, V. E. Calado, G. Pandraud, H. W. Zandbergen, L. M. K. Vandersypen and C. Dekker, *Nano Lett.*, 2010, **10**, 3163–3167.
- 53 C. J. Russo and J. A. Golovchenko, *Proc. Natl. Acad. Sci.*, 2012, **109**, 5953.
- 54 H. Wang, K. Kurata, T. Fukunaga, H. Takamatsu, X. Zhang, T. Ikuta, K. Takahashi, T. Nishiyama, H. Ago and Y. Takata, *Carbon*, 2016, **99**, 564–570.
- 55 K. Celebi, J. Buchheim, R. M. Wyss, A. Droudian, P. Gasser, I. Shorubalko, J.-I. Kye, C. Lee and H. G. Park, *Science*, 2014, **344**, 289.
- 56 S. C. O'Hern, M. S. H. Boutilier, J.-C. Idrobo, Y. Song, J. Kong, T. Laoui, M. Atieh and R. Karnik, *Nano Lett.*, 2014, **14**, 1234–1241.
- 57 H. Vázquez, E. H. Åhlgren, O. Ochedowski, A. A. Leino, R. Mirzayev, R. Kozubek, H. Lebius, M. Karlušić, M. Jakšić, A. V. Krasheninnikov, J. Kotakoski, M. Schleberger, K. Nordlund and F. Djurabekova, *Carbon*, 2017, **114**, 511–518.
- 58 D. K. Avasthi, *Curr. Sci.*, 2000, **78**, 1297–1303.
- 59 N. A. Nebogatikova, I. V. Antonova, S. V. Erohin, D. G. Kvashnin, A. Olejniczak, V. A. Volodin, A. V. Skuratov, A. V. Krasheninnikov, P. B. Sorokin and L. A. Chernozatonskii, *Nanoscale*, 2018, **10**, 14499–14509.
- 60 W. Li, L. Liang, S. Zhao, S. Zhang and J. Xue, *J. Appl. Phys.*, 2013, **114**, 234304.
- 61 S. Garaj, S. Liu, J. A. Golovchenko and D. Branton, *Proc. Natl. Acad. Sci.*, 2013, **110**, 12192.
- 62 J. Buchheim, R. M. Wyss, I. Shorubalko and H. G. Park, *Nanoscale*, 2016, **8**, 8345–8354.
- 63 Y.-Y. Peng, Y.-M. Liu, J.-K. Chang, C.-H. Wu, M.-D. Ger, N.-W. Pu and C.-L. Chang, *Carbon*, 2015, **81**, 347–356.



- 64 C.-H. Wu, N.-W. Pu, Y.-M. Liu, C.-Y. Chen, Y.-Y. Peng, T.-Y. Cheng, M.-H. Lin and M.-D. Ger, *J. Taiwan Inst. Chem. Eng.*, 2017, **80**, 511–517.
- 65 F. Su, S. Zheng, F. Liu, X. Zhang, F. Su and Z.-S. Wu, *Chin. Chem. Lett.*, 2021, **32**, 914–917.
- 66 C.-H. Yang, P.-L. Huang, X.-F. Luo, C.-H. Wang, C. Li, Y.-H. Wu and J.-K. Chang, *ChemSusChem*, 2015, **8**, 1779–1786.
- 67 B. Xing, H. Zeng, G. Huang, C. Zhang, R. Yuan, Y. Cao, Z. Chen and J. Yu, *J. Alloys Compd.*, 2019, **779**, 202–211.
- 68 D. Zhou, Q.-Y. Cheng and B.-H. Han, *Carbon*, 2011, **49**, 3920–3927.
- 69 X.-F. Luo, C.-H. Yang and J.-K. Chang, *J. Mater. Chem. A*, 2015, **3**, 17282–17289.
- 70 Y.-R. Huang, C.-L. Chen, N.-W. Pu, C.-H. Wu, Y.-M. Liu, Y.-H. Chen, M.-J. Youh and M.-D. Ger, *Crystals*, 2020, **10**, 1063.
- 71 X. Han, M. R. Funk, F. Shen, Y.-C. Chen, Y. Li, C. J. Campbell, J. Dai, X. Yang, J.-W. Kim, Y. Liao, J. W. Connell, V. Barone, Z. Chen, Y. Lin and L. Hu, *ACS Nano*, 2014, **8**, 8255–8265.
- 72 Y. Lin, X. Han, C. J. Campbell, J.-W. Kim, B. Zhao, W. Luo, J. Dai, L. Hu and J. W. Connell, *Adv. Funct. Mater.*, 2015, **25**, 2920–2927.
- 73 Y. Jin, C. Hu, Q. Dai, Y. Xiao, Y. Lin, J. W. Connell, F. Chen and L. Dai, *Adv. Funct. Mater.*, 2018, **28**, 1804630.
- 74 J. Xu, Y. Lin, J. W. Connell and L. Dai, *Small*, 2015, **11**, 6179–6185.
- 75 Y. Chen, Y. Wang, S. Zhu, K. Fu, X. Han, Y. Wang, B. Zhao, T. Li, B. Liu, Y. Li, J. Dai, H. Xie, T. Li, J. W. Connell, Y. Lin and L. Hu, *Mater. Today*, 2019, **24**, 26–32.
- 76 F. Xiao, X. Chen, J. Zhang, C. Huang, T. Hu, B. Hong and J. Xu, *J. Energy Chem.*, 2020, **48**, 122–127.
- 77 Y. Lin, K. A. Watson, J.-W. Kim, D. W. Baggett, D. C. Working and J. W. Connell, *Nanoscale*, 2013, **5**, 7814–7824.
- 78 E. Alsharaeh, F. Ahmed, Y. Aldawsari, M. Khasawneh, H. Abuhimad and M. Alshahrani, *Sci. Rep.*, 2016, **6**, 29854.
- 79 W. Seo, D. L. White and A. Star, *Chem.–Eur. J.*, 2017, **23**, 5652–5657.
- 80 J. Wang, T. Park, J. W. Yi, B. Ding, J. Henzie, Z. Chang, H. Dou, X. Zhang and Y. Yamauchi, *Nanoscale Horiz.*, 2019, **4**, 526–530.
- 81 Z. Du, W. Ai, C. Sun, C. Zou, J. Zhao, Y. Chen, X. Dong, J. Liu, G. Sun, T. Yu and W. Huang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 33712–33722.
- 82 D. Dutta, J.-Y. Jiang, A. Jamaluddin, S.-M. He, Y.-H. Hung, F. Chen, J.-K. Chang and C.-Y. Su, *ACS Appl. Mater. Interfaces*, 2019, **11**, 36560–36570.
- 83 X. Zhao, C. M. Hayner, M. C. Kung and H. H. Kung, *ACS Nano*, 2011, **5**, 8739–8749.
- 84 X. Wang, L. Jiao, K. Sheng, C. Li, L. Dai and G. Shi, *Sci. Rep.*, 2013, **3**, 1996.
- 85 J.-H. Chang, Y.-H. Hung, X.-F. Luo, C.-H. Huang, S. Jung, J.-K. Chang, J. Kong and C.-Y. Su, *RSC Adv.*, 2016, **6**, 8384–8394.
- 86 S. Chowdhury and R. Balasubramanian, *Sci. Rep.*, 2016, **6**, 21537.
- 87 Y. Xu, C.-Y. Chen, Z. Zhao, Z. Lin, C. Lee, X. Xu, C. Wang, Y. Huang, M. I. Shakir and X. Duan, *Nano Lett.*, 2015, **15**, 4605–4610.
- 88 Y. Xu, Z. Lin, X. Zhong, X. Huang, N. O. Weiss, Y. Huang and X. Duan, *Nat. Commun.*, 2014, **5**, 4554.
- 89 J. Lv and T. Liang, *Chem. Phys. Lett.*, 2016, **659**, 61–65.
- 90 W. Kong, X. Duan, Y. Ge, H. Liu, J. Hu and X. Duan, *Nano Res.*, 2016, **9**, 2458–2466.
- 91 T. Palaniselvam, M. O. Valappil, R. Illathvalappil and S. Kurungot, *Energy Environ. Sci.*, 2014, **7**, 1059–1067.
- 92 C. Xu, X. He, C. Wang, X. Chen, R. Yuan and W. Dai, *RSC Adv.*, 2016, **6**, 84068–84073.
- 93 D. Wang, R. Dai, X. Zhang, L. Liu, H. Zhuang, Y. Lu, Y. Wang, Y. Liao and Q. Nian, *Carbon*, 2020, **161**, 880–891.
- 94 D. C. Marcano, D. V. Kosynkin, J. M. Berlin, A. Sinitskii, Z. Sun, A. Slesarev, L. B. Alemany, W. Lu and J. M. Tour, *ACS Nano*, 2010, **4**, 4806–4814.
- 95 M. A. Lillo-Ródenas, D. Cazorla-Amorós and A. Linares-Solano, *Carbon*, 2003, **41**, 267–275.
- 96 Y. Zhu, S. Murali, M. D. Stoller, K. J. Ganesh, W. Cai, P. J. Ferreira, A. Pirkle, R. M. Wallace, K. A. Cychoz, M. Thommes, D. Su, E. A. Stach and R. S. Ruoff, *Science*, 2011, **332**, 1537.
- 97 L. Chunlan, X. Shaoping, G. Yixiong, L. Shuqin and L. Changhou, *Carbon*, 2005, **43**, 2295–2301.
- 98 L. Xu, W. Lv, K. Shi, S. Xiao, C. You, Y.-B. He, F. Kang and Q.-H. Yang, *Carbon*, 2019, **149**, 257–262.
- 99 K. Xia, Q. Li, L. Zheng, K. You, X. Tian, B. Han, Q. Gao, Z. Huang, G. Chen and C. Zhou, *Microporous Mesoporous Mater.*, 2017, **237**, 228–236.
- 100 C. Zheng, X. F. Zhou, H. L. Cao, G. H. Wang and Z. P. Liu, *J. Mater. Chem. A*, 2015, **3**, 9543–9549.
- 101 I. A. Baburin, A. Klechikov, G. Mercier, A. Talyzin and G. Seifert, *Int. J. Hydrogen Energy*, 2015, **40**, 6594–6599.
- 102 S. Wu, K. S. Hui, K. N. Hui, J. M. Yun and K. H. Kim, *Chem. Eng. J.*, 2017, **317**, 461–470.
- 103 Z. Jiang, B. Pei and A. Manthiram, *J. Mater. Chem. A*, 2013, **1**, 7775–7781.
- 104 Z. Zhang, T. Cao, S. Liu, X. Duan, L.-M. Liu, S. Wang and Y. Liu, *Part. Part. Syst. Charact.*, 2017, **34**, 1600207.
- 105 G. Ning, Z. Fan, G. Wang, J. Gao, W. Qian and F. Wei, *Chem. Commun.*, 2011, **47**, 5976–5978.
- 106 G. Ning, C. Xu, L. Hao, O. Kazakova, Z. Fan, H. Wang, K. Wang, J. Gao, W. Qian and F. Wei, *Carbon*, 2013, **51**, 390–396.
- 107 A. Sinitskii and J. M. Tour, *J. Am. Chem. Soc.*, 2010, **132**, 14730–14732.
- 108 M. Wang, L. Fu, L. Gan, C. Zhang, M. Rummeli, A. Bachmatiuk, K. Huang, Y. Fang and Z. Liu, *Sci. Rep.*, 2013, **3**, 1238.
- 109 J.-L. Shi, H.-J. Peng, L. Zhu, W. Zhu and Q. Zhang, *Carbon*, 2015, **92**, 96–105.
- 110 C. Tang, B.-Q. Li, Q. Zhang, L. Zhu, H.-F. Wang, J.-L. Shi and F. Wei, *Adv. Funct. Mater.*, 2016, **26**, 577–585.



- 111 R. P. Panmand, P. Patil, Y. Sethi, S. R. Kadam, M. V. Kulkarni, S. W. Gosavi, N. R. Munirathnam and B. B. Kale, *Nanoscale*, 2017, **9**, 4801–4809.
- 112 H. Zhu, X. Wang, F. Yang and X. Yang, *Adv. Mater.*, 2011, **23**, 2745–2748.
- 113 Y. Wang, T. Song, P. Zhang, T. Huang, T. Wang, T. Wang and H. Zeng, *ACS Sustainable Chem. Eng.*, 2018, **6**, 11536–11546.
- 114 A.-K. Lu, H.-Y. Li and Y. Yu, *J. Mater. Chem. A*, 2019, **7**, 7852–7858.
- 115 X. Liang, Y.-S. Jung, S. Wu, A. Ismach, D. L. Olynick, S. Cabrini and J. Bokor, *Nano Lett.*, 2010, **10**, 2454–2460.
- 116 H. Yang, C. Zhang, Q. Meng, B. Cao and G. Tian, *J. Power Sources*, 2019, **431**, 114–124.
- 117 F. Wang, X. Mei, K. Wang, X. Dong, M. Gao, Z. Zhai, J. Lv, C. Zhu, W. Duan and W. Wang, *J. Mater. Sci.*, 2019, **54**, 5658–5670.
- 118 Z. Li, X. Zhang, H. Tan, W. Qi, L. Wang, M. C. Ali, H. Zhang, J. Chen, P. Hu, C. Fan and H. Qiu, *Adv. Funct. Mater.*, 2018, **28**, 1805026.
- 119 C. Moreno, M. Vilas-Varela, B. Kretz, A. Garcia-Lekue, M. V. Costache, M. Paradinas, M. Panighel, G. Ceballos, S. O. Valenzuela, D. Peña and A. Mugarza, *Science*, 2018, **360**, 199.
- 120 A. Al-Hagri, R. Li, N. S. Rajput, J.-Y. Lu, S. Cong, K. Sloyan, M. A. Almahri, S. R. Tamalampudi, M. Chiesa and A. Al Ghaferi, *Carbon*, 2019, **155**, 320–325.
- 121 X. J. Lee, B. Y. Z. Hiew, K. C. Lai, L. Y. Lee, S. Gan, S. Thangalazhy-Gopakumar and S. Rigby, *J. Taiwan Inst. Chem. Eng.*, 2019, **98**, 163–180.
- 122 C. Zhu, Z. Hui, H. Pan, S. Zhu, Q. Zhang, J. Mao, Z. Guo, Y. Li, M. Imtiaz and Z. Chen, *J. Mater. Chem. A*, 2019, **7**, 4788–4796.
- 123 Y. Lin, B. Moitoso, C. Martinez-Martinez, E. D. Walsh, S. D. Lacey, J.-W. Kim, L. Dai, L. Hu and J. W. Connell, *Nano Lett.*, 2017, **17**, 3252–3260.
- 124 J. Yang, X. Shan, Z. Guo, L. Duan, X. Zhang and W. Lü, *J. Electroanal. Chem.*, 2020, **876**, 114728.
- 125 J. H. Jeong, G.-W. Lee, Y. H. Kim, Y. J. Choi, K. C. Roh and K.-B. Kim, *Chem. Eng. J.*, 2019, **378**, 122126.
- 126 D. Yang, B. Xu, Q. Zhao and X. S. Zhao, *J. Mater. Chem. A*, 2019, **7**, 363–371.
- 127 Y. Chen, Y. Li, F. Yao, C. Peng, C. Cao, Y. Feng and W. Feng, *Sustainable Energy Fuels*, 2019, **3**, 2237–2245.
- 128 D. B. Wells, M. Belkin, J. Comer and A. Aksimentiev, *Nano Lett.*, 2012, **12**, 4117–4123.
- 129 Y.-s. Yu, X. Lu, H.-m. Ding and Y.-q. Ma, *Phys. Chem. Chem. Phys.*, 2018, **20**, 9063–9069.
- 130 A. Wasfi, F. Awwad and A. I. Ayesh, *Biosens. Bioelectron.*, 2018, **119**, 191–203.
- 131 S. J. Heerema and C. Dekker, *Nat. Nanotechnol.*, 2016, **11**, 127–136.
- 132 H. W. C. Postma, *Nano Lett.*, 2010, **10**, 420–425.
- 133 S. Garaj, W. Hubbard, A. Reina, J. Kong, D. Branton and J. A. Golovchenko, *Nature*, 2010, **467**, 190–193.
- 134 C. A. Merchant, K. Healy, M. Wanunu, V. Ray, N. Peterman, J. Bartel, M. D. Fischbein, K. Venta, Z. Luo, A. T. C. Johnson and M. Drndić, *Nano Lett.*, 2010, **10**, 2915–2921.
- 135 J. Lee, Z. Yang, W. Zhou, S. J. Pennycook, S. T. Pantelides and M. F. Chisholm, *Proc. Natl. Acad. Sci.*, 2014, **111**, 7522.
- 136 C. Buelke, A. Alshami, J. Casler, J. Lewis, M. Al-Sayaghi and M. A. Hickner, *Desalination*, 2018, **448**, 113–132.
- 137 C. Shao, Y. Zhao and L. Qu, *ChemNanoMat*, 2020, **6**, 1028–1048.
- 138 X. Li, B. Zhu and J. Zhu, *Carbon*, 2019, **146**, 320–328.
- 139 M. Zahed, P. S. Parsamehr, M. A. Tofighy and T. Mohammadi, *Chem. Eng. Res. Des.*, 2018, **138**, 358–365.
- 140 H. M. Hegab and L. Zou, *J. Membr. Sci.*, 2015, **484**, 95–106.
- 141 C.-N. Yeh, K. Raidongia, J. Shao, Q.-H. Yang and J. Huang, *Nat. Chem.*, 2015, **7**, 166–170.
- 142 K. Hatakeyama, K. Awaya, M. Koinuma, Y. Shimizu, Y. Hakuta and Y. Matsumoto, *Soft Matter*, 2017, **13**, 8353–8356.
- 143 F. Dai, R. Yu, R. Yi, J. Lan, R. Yang, Z. Wang, J. Chen and L. Chen, *Chem. Commun.*, 2020, **56**, 15068–15071.
- 144 Y. Shin, W. Liu, B. Schwenzer, S. Manandhar, D. Chase-Woods, M. H. Engelhard, R. Devanathan, L. S. Fifield, W. D. Bennett, B. Ginovska and D. W. Gotthold, *Carbon*, 2016, **106**, 164–170.
- 145 N. Wei, X. Peng and Z. Xu, *ACS Appl. Mater. Interfaces*, 2014, **6**, 5877–5883.
- 146 Z. Xie, Y. Duo, Z. Lin, T. Fan, C. Xing, L. Yu, R. Wang, M. Qiu, Y. Zhang, Y. Zhao, X. Yan and H. Zhang, *Adv. Sci.*, 2020, **7**, 1902236.
- 147 Z. Xie, Y.-P. Peng, L. Yu, C. Xing, M. Qiu, J. Hu and H. Zhang, *Sol. RRL*, 2020, **4**, 1900400.
- 148 Y. Yang, R. Zhao, T. Zhang, K. Zhao, P. Xiao, Y. Ma, P. M. Ajayan, G. Shi and Y. Chen, *ACS Nano*, 2018, **12**, 829–835.
- 149 Z. Chen, Q. Li and X. Chen, *ACS Sustainable Chem. Eng.*, 2020, **8**, 13850–13858.
- 150 Y. Ito, Y. Tanabe, J. Han, T. Fujita, K. Tanigaki and M. Chen, *Adv. Mater.*, 2015, **27**, 4302–4307.
- 151 L. Wang, Z. Li, J. Chen, Y. Huang, H. Zhang and H. Qiu, *Environ. Pollut.*, 2019, **249**, 801–811.
- 152 Z. Bin and L. Hui, *Appl. Surf. Sci.*, 2015, **357**, 439–444.
- 153 R. Nagar, B. P. Vinayan, S. S. Samantaray and S. Ramaprabhu, *J. Mater. Chem. A*, 2017, **5**, 22897–22912.
- 154 H. Tachikawa and T. Iyama, *J. Phys. Chem. C*, 2019, **123**, 8709–8716.
- 155 S. Huang, S. Li, L. F. Villalobos, M. Dakhchoune, M. Micari, D. J. Babu, M. T. Vahdat, M. Mensi, E. Oveisi and K. V. Agrawal, *Sci. Adv.*, 2021, **7**, eabf0116.
- 156 G. He, S. Huang, L. F. Villalobos, J. Zhao, M. Mensi, E. Oveisi, M. Rezaei and K. V. Agrawal, *Energy Environ. Sci.*, 2019, **12**, 3305–3312.
- 157 C. Chakravarty, B. Mandal and P. Sarkar, *J. Phys. Chem. C*, 2018, **122**, 15835–15842.
- 158 Z. Gao, L. Wang, J. Chang, X. Liu, D. Wu, F. Xu, Y. Guo and K. Jiang, *Electrochim. Acta*, 2016, **188**, 441–449.
- 159 J. Bai, X. Zhong, S. Jiang, Y. Huang and X. Duan, *Nat. Nanotechnol.*, 2010, **5**, 190–194.





## Review

- 160 Z. Jiang, Y. Shi, Z.-J. Jiang, X. Tian, L. Luo and W. Chen, *J. Mater. Chem. A*, 2014, **2**, 6494–6503.
- 161 C.-Y. Chen, N.-W. Pu, Y.-M. Liu, L.-H. Chen, C.-H. Wu, T.-Y. Cheng, M.-H. Lin, M.-D. Ger, Y.-J. Gong, Y.-Y. Peng, P. M. Grubb and R. T. Chen, *Composites, Part B*, 2018, **135**, 119–128.
- 162 Q. Li, X. Tian, W. Yang, L. Hou, Y. Li, B. Jiang, X. Wang and Y. Li, *Appl. Surf. Sci.*, 2020, **530**, 147298.
- 163 Y. Zheng, X. Wang, S. Wei, B. Zhang, M. Yu, W. Zhao and J. Liu, *Composites, Part A*, 2017, **95**, 237–247.
- 164 C. Chen, J. Xi, E. Zhou, L. Peng, Z. Chen and C. Gao, *Nano-Micro Lett.*, 2017, **10**, 26.
- 165 G. Deokar, J. Casanova-Cháfer, N. S. Rajput, C. Aubry, E. Llobet, M. Jouiad and P. M. F. J. Costa, *Sens. Actuators, B*, 2020, **305**, 127458.
- 166 T. Wang, D. Huang, Z. Yang, S. Xu, G. He, X. Li, N. Hu, G. Yin, D. He and L. Zhang, *Nano-Micro Lett.*, 2016, **8**, 95–119.
- 167 S. Rumyantsev, G. Liu, M. S. Shur, R. A. Potyrailo and A. A. Balandin, *Nano Lett.*, 2012, **12**, 2294–2298.
- 168 Y. Wang, M. Yang, W. Liu, L. Dong, D. Chen and C. Peng, *J. Mater. Chem. C*, 2019, **7**, 9248–9256.
- 169 D. H. Wang, Y. Hu, J. J. Zhao, L. L. Zeng, X. M. Tao and W. Chen, *J. Mater. Chem. A*, 2014, **2**, 17415–17420.
- 170 G. Li, D. Dong, G. Hong, L. Yan, X. Zhang and W. Song, *Adv. Mater.*, 2019, **31**, 1901403.
- 171 J. S. Lee, D. K. Seo, T. Kim and Y. H. Kim, *J. Mater. Chem. C*, 2019, **7**, 1131–1137.
- 172 Y. Chen, Z. P. Michael, G. P. Kotchey, Y. Zhao and A. Star, *ACS Appl. Mater. Interfaces*, 2014, **6**, 3805–3810.
- 173 O. Akhavan and E. Ghaderi, *Small*, 2013, **9**, 3593–3601.
- 174 K. Khan, A. K. Tareen, M. Aslam, R. Wang, Y. Zhang, A. Mahmood, Z. Ouyang, H. Zhang and Z. Guo, *J. Mater. Chem. C*, 2020, **8**, 387–440.
- 175 J. Pei, J. Yang, T. Yildirim, H. Zhang and Y. Lu, *Adv. Mater.*, 2019, **31**, 1706945.
- 176 M. Zhang, Q. Wu, F. Zhang, L. Chen, X. Jin, Y. Hu, Z. Zheng and H. Zhang, *Adv. Opt. Mater.*, 2019, **7**, 1800224.
- 177 X. Qi, Y. Zhang, Q. Ou, S. T. Ha, C.-W. Qiu, H. Zhang, Y.-B. Cheng, Q. Xiong and Q. Bao, *Small*, 2018, **14**, 1800682.
- 178 B. Guo, Q.-l. Xiao, S.-h. Wang and H. Zhang, *Laser Photonics Rev.*, 2019, **13**, 1800327.
- 179 X. Jiang, A. V. Kuklin, A. Baev, Y. Ge, H. Ågren, H. Zhang and P. N. Prasad, *Phys. Rep.*, 2020, **848**, 1–58.
- 180 E. S. Polsen, D. Q. McNerny, B. Viswanath, S. W. Pattinson and A. John Hart, *Sci. Rep.*, 2015, **5**, 10257.
- 181 A. Alnuaimi, I. Almansouri, I. Saadat and A. Nayfeh, *RSC Adv.*, 2017, **7**, 51951–51957.
- 182 Y. Zhang, L. Zhang and C. Zhou, *Acc. Chem. Res.*, 2013, **46**, 2329–2339.

