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# The site pair matching of a tandem Au/CuO–CuO nanocatalyst for promoting the selective electrolysis of CO<sub>2</sub> to C<sub>2</sub> products†

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Tandem catalysis, in which a CO<sub>2</sub>-to-C<sub>2</sub> process is divided into a CO<sub>2</sub>-to-CO/\*CO step and a CO/\*CO-to-C<sub>2</sub> step, is promising for enhancing the C<sub>2</sub> product selectivity when using Cu-based electrochemical CO<sub>2</sub> reduction catalysts. In this work, a nanoporous hollow Au/CuO–CuO tandem catalyst was used for catalyzing the eCO<sub>2</sub>RR, which exhibited a C<sub>2</sub> product FE of 52.8% at –1.0 V vs. RHE and a C<sub>2</sub> product partial current density of 78.77 mA cm<sup>–2</sup> at –1.5 V vs. RHE. In addition, the C<sub>2</sub> product FE stably remained at over 40% over a wide potential range, from –1.0 V to –1.5 V. This superior performance was attributed to good matching in terms of the optimal working potential and charge-transfer resistance between CO/\*CO-production sites (Au/CuO) and CO/\*CO-reduction sites (CuO). This site pair matching effect ensured sufficient supplies of CO/\*CO and electrons at CuO sites at the working potentials, thus dramatically enhancing the formation rate of C<sub>2</sub> products.

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## Introduction

The electrochemical CO<sub>2</sub> reduction reaction (eCO<sub>2</sub>RR), which can convert CO<sub>2</sub> into valuable products, is a promising technique for lowering the concentration of atmospheric CO<sub>2</sub>, recycling and reusing CO<sub>2</sub> to offer carbon-neutral chemical feedstocks, and storing green electricity generated using renewable energy such as from wind and solar sources.<sup>1–3</sup> In recent years, enormous progress has been achieved in terms of catalyst design to promote the two-electron reduction of CO<sub>2</sub> to produce 2e<sup>–</sup> products (CO or formate).<sup>4–8</sup> However, 2e<sup>–</sup> products are much less attractive due to their limited market size and low market price.<sup>9</sup> Multi-carbon products (C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>5</sub>OH, CH<sub>3</sub>COOH, *etc.*), which are more valuable and have higher energy concentrations, have drawn widespread attention recently.<sup>10–14</sup> Theoretical and experimental evidence has illustrated that CO<sub>2</sub> is first adsorbed on the surface of the electrode in the form of \*COOH, and it is then converted into \*CO, followed by further reduction to hydrocarbons and oxygenates.<sup>15–17</sup> C–C coupling and the formation of C<sub>2</sub> products can only happen when the \*CO intermediate is moderately adsorbed on the surface of the electrode, and Cu surfaces have the optimal binding

strength for \*CO.<sup>18,19</sup> As a result, Cu-based electrocatalysts are almost the only materials capable of catalyzing the reduction of CO<sub>2</sub> to C<sub>2</sub> products.

Therefore, various catalyst design principles have been implemented to improve the catalytic activity and selectivity for C<sub>2</sub> products of Cu-based electrocatalysts.<sup>20</sup> The most common approach involves alloying Cu with a heterometal to tune the binding strengths of specific intermediates, using oxidation and reduction treatments to modify Cu surfaces, utilizing morphology control to expose surfaces with abundant low-coordinated sites, and so on.<sup>21–25</sup> However, the improvements in catalytic performance remain unsatisfactory, and this can mostly be attributed to the fact that simultaneously optimizing the binding strengths of \*COOH and \*CO on one type of reactive site of a Cu-based electrocatalyst is quite challenging.<sup>26</sup> As a result, tandem catalysis, in which the CO<sub>2</sub>-to-C<sub>2</sub> process is divided into a CO<sub>2</sub>-to-CO/\*CO step and a CO/\*CO-to-C<sub>2</sub> step, and the formation of CO/\*CO and the reduction of CO/\*CO are catalyzed by different active sites, thus modifying the binding strengths of \*COOH and \*CO simultaneously, is much easier, making it a promising approach for further promoting the formation of C<sub>2</sub> products.<sup>27</sup> Several kinds of tandem eCO<sub>2</sub>RR catalysts have been reported very recently. Such catalysts usually consist of CO/\*CO-production sites (for example, Ag or Au, single-atom catalysts, complexes) and CO/\*CO-reduction sites (usually Cu).<sup>28–30</sup> For example, Yang's group reported a Cu–Ag tandem catalyst which could achieve a 4-fold enhancement in the multi-carbon formation rate, while Jaramillo's group found that a tandem Au-on-Cu electrocatalyst was over 100 times more selective for the formation of C<sub>2+</sub> products over methane or methanol.<sup>31,32</sup>

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When it comes to Au–Cu systems, many excellent studies have obtained high  $C_{2+}$  product selectivity, and the reasons for these performance improvements have been explained. Zhang's group developed a general strategy for synthesizing Au–Cu Janus nanocrystals, and the obtained catalyst attained 46.4% faradaic efficiency (FE) in relation to  $C_2$  products at  $-0.98$  V vs. reverse hydrogen electrode (RHE) (all potentials hereafter are provided with respect to RHE, unless otherwise stated).<sup>33</sup> Grätzel's group designed CuAu bimetallic catalysts through galvanic exchange between  $Au^{3+}$  and  $Cu_2O$  nanowires, and 70.1% FE with respect to  $C_{2+}$  products was obtained at  $-1.05$  V.<sup>34</sup> Altogether, these excellent electrocatalytic performances have been attributed to appropriate spatial distances and high local  $*CO$  coverages at interfaces.

However, in all of these studies, researchers only focused on fabricating tandem catalysts, and they did not pay attention to the conditions under which tandem catalysis could occur and proceed most effectively and efficiently. In a significant number of publications reporting Au–Cu and Ag–Cu  $eCO_2RR$  catalysts, tandem catalysis did not occur at all, and  $2e^-$  products were almost exclusively produced.<sup>35–38</sup> Therefore, merely combining components with  $CO/*CO$ -production abilities and  $CO/*CO$ -reduction abilities does not necessarily result in a tandem  $eCO_2RR$  catalyst. There must be some conditions that limit the occurrence of a tandem catalysis process. If we can solve this problem, we could develop tandem catalysts with satisfactory performance more predictably.

Herein, we synthesized a nanoporous hollow Au/CuO–CuO tandem catalyst *via* annealing Au/Cu<sub>2</sub>O<sub>4</sub>/C in air at 400 °C. The Au/CuO–CuO tandem catalyst exhibited a  $C_2$  FE of 52.8% at  $-1.0$  V and a  $C_2$  partial current density of 78.77 mA cm<sup>-2</sup> at  $-1.5$  V. Furthermore, the  $C_2$  FE was stably maintained at above 40% over a wide potential range from  $-1.0$  V to  $-1.5$  V. These results represent a significant improvement over a CuO catalyst synthesized through a similar procedure, and the electrochemical CO reduction reaction (eCORR) results could confirm the existence of a tandem catalysis process. In particular, we observed that a catalyst obtained *via* the physical mixing of Au and CuO catalysts prepared under the same procedure only produces  $C_1$  products, providing an ingenious comparison method for inferring the required conditions for the occurrence of a tandem reaction. More experiments elucidate the conditions required for tandem catalysis. The  $CO/*CO$ -production sites (Au/CuO) and  $CO/*CO$ -reduction sites (CuO) matched well in terms of the optimal working potential and charge-transfer resistance, ensuring that a sufficient amount of  $CO/*CO$  was produced and that there was a sufficient electron supply at CuO sites at potentials more negative than  $-1.0$  V, thus further promoting the formation rate of  $C_2$  products to a satisfactory level.

## Experimental section

### Chemicals

H<sub>2</sub>AuCl<sub>4</sub>·4H<sub>2</sub>O (A.R., Beijing Chemical Works, China), CuCl<sub>2</sub>·2H<sub>2</sub>O (≥99%, Xilong Scientific Co., Ltd, China), Na<sub>2</sub>C<sub>2</sub>O<sub>4</sub> (A.R., Beijing Chemical Works, China), HCl solution (36%, A.R., Xilong Scientific Co., Ltd, China), KOH (≥99.999%, Aladdin),

IrO<sub>2</sub> (≥85.5%, Sun Chemical Technology (Shanghai) Co., Ltd, China), cupric oxide (99.5%, particle size: 100–200 nm, Macklin), isopropanol (A.R., Xilong Scientific Co., Ltd, China), Nafion perfluorinated resin solution (5 wt% in lower aliphatic alcohols and water, Sigma-Aldrich), D<sub>2</sub>O (A.R., Sigma-Aldrich), dimethyl sulfoxide (≥99.8%, Aladdin), and deionized water (Millipore, 18.2 MΩ cm) were obtained from the listed sources.

N<sub>2</sub> (99.99%) and compressed air (extra dry) were purchased from Beijing Haike Yuanchang Practical Gases Co., Ltd; H<sub>2</sub> (99.999%) was purchased from Beijing AP BAIF Gases Industry Co., Ltd; CO<sub>2</sub> (99.999%) was purchased from Beijing Beiwen Gases Co., Ltd; and CO (99.9%) was purchased from Beijing Nanfei Gases Co., Ltd.

### Synthesis of Au/Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles, Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles, and Au/C nanoparticles

Au/Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles were synthesized through a one-pot method, including a co-precipitation step and reduction step. Specifically, 240 mg of carbon black was dispersed in 20 mL of deionized water in a 100 mL flask, followed by the addition of 1 mL of HCl solution (36%). Then, 2 mmol of CuCl<sub>2</sub>·2H<sub>2</sub>O was added into the flask under vigorous stirring. After 10 min, 30 mL of 0.1 M Na<sub>2</sub>C<sub>2</sub>O<sub>4</sub> aqueous solution was added into the flask and the mixture was kept under stirring for 60 min. Then the solution was heated to 80 °C and 5 mL of 0.02 M HAuCl<sub>4</sub> aqueous solution was introduced into the flask. The mixture was kept at 80 °C for 4 h. The product was collected *via* centrifugation at 9500 rpm for 10 min and washed repeatedly with deionized water using a redispersion–centrifugation method, followed by drying at 50 °C overnight.

Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles were synthesized through a similar procedure, except that after the mixture was heated to 80 °C, no HAuCl<sub>4</sub> precursor was added.

Au/C nanoparticles were synthesized through a similar procedure, except that at the beginning of synthesis, no CuCl<sub>2</sub> precursor was added.

### Annealing of Au/Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles, Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles, and Au/C nanoparticles

In a typical procedure, appropriate amounts of as-synthesized Au/Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles, Cu<sub>2</sub>O<sub>4</sub>/C nanoparticles, and Au/C nanoparticles were annealed at 400 °C for 4 h in air at a heating rate of 10 °C min<sup>-1</sup>, individually, and then cooled down to room temperature. The obtained samples were named AuCu400, Cu400, and Au400, respectively.

### Synthesis of Au400 + Cu400 and Au400 + NPCuO samples

Appropriate amounts of Au400 and Cu400 were mixed together, and the obtained sample was named Au400 + Cu400. Appropriate amounts of Au400, commercial CuO NPs (NPCuO), and carbon black were mixed together, and the obtained sample was named Au400 + NPCuO. The amounts of Au400, Cu400, and NPCuO was weighed precisely based on the ratio of Cu and Au in AuCu400 obtained from ICP-OES results, so the ratios of Cu and Au in both Au400 + Cu400 and Au400 + NPCuO were nearly the same as that in AuCu400.



### Preparation of working electrodes

In this step, AuCu400, Cu400, Au400, AuCu (electrodes containing Au/Cu<sub>2</sub>O<sub>4</sub>/C), Au400 + Cu400, Au400 + NPCuO, and NPCuO were treated in the same way. For convenience, we take AuCu400 as an example. In a typical procedure, 5 mg of AuCu400 powder was dispersed in 1000  $\mu$ L of 0.5 wt% Nafion solution *via* sonication for 1 h to form a homogeneous ink. The mixture was then sprayed onto carbon paper with a size of 2 cm  $\times$  0.5 cm with a gas diffusion layer and dried using a heating plate, yielding the working electrode.

### Preparation of counter electrodes

In a typical procedure, 2.5 mg of purchased IrO<sub>2</sub> powder was dispersed in 1000  $\mu$ L of 0.5 wt% Nafion solution *via* sonication for 1 h to form a homogeneous ink. The mixture was then sprayed onto carbon paper with a size of 2 cm  $\times$  0.5 cm with a gas diffusion layer and dried using a heating plate, yielding the counter electrode.

### Electrochemical measurements and product analysis

All electrolysis experiments involving CO<sub>2</sub> and CO were conducted in a gas diffusion flow cell. In this work, 1 M KOH was used as the electrolyte.<sup>39–41</sup> The gas flow rate during electrolysis was controlled using a mass flow controller at 10 sccm, while the electrolyte flow rates in the cathode and anode channels were managed using a peristaltic pump at 5 mL min<sup>-1</sup>. The out-flowing electrolyte was fed back to the cell circularly. During all electrolysis experiments, the cathode and anode were separated by a Fumasep anion exchange membrane. Ag/AgCl (saturated KCl) was used as the reference electrode. During chronoamperometry, effluent gas from the cell was introduced directly to the gas sample loop of a gas chromatograph (GC), which was equipped with two Porapak-N MS-13X and HP-AL/S columns for quantification. The gas-phase products were analyzed *via* GC every 13 min. CO in the gas products was analyzed using a thermal conductivity detector (TCD) with flowing H<sub>2</sub> as a carrier gas, H<sub>2</sub> in the gas products was analyzed using another TCD with flowing N<sub>2</sub> as a carrier gas, and hydrocarbons in the gas products were analyzed using a flame ionization detector (FID) with flowing N<sub>2</sub> as a carrier gas. The liquid products were characterized *via* <sup>1</sup>H nuclear magnetic resonance (NMR) spectroscopy. 0.5 mL of electrolyte, after electrolysis, was mixed with 0.1 mL of D<sub>2</sub>O, and 0.05  $\mu$ L of dimethyl sulfoxide (DMSO) was added as an internal standard. One-dimensional <sup>1</sup>H spectra were measured with water suppression, implementing a solvent pre-saturation technique.

Linear sweep voltammetry (LSV) was carried out at a scan rate of 5 mV s<sup>-1</sup>.

To rule out the possibility that C-based products could be produced *via* the decomposition of electrolyte (CO<sub>2</sub> reacts with KOH to form K<sub>2</sub>CO<sub>3</sub> or KHCO<sub>3</sub>), electrolysis with AuCu400 using N<sub>2</sub> as the feed gas was performed.

The electrochemical impedance spectroscopy (EIS) analysis of samples was carried out using an AC voltage with an amplitude of 5 mV at  $-1.0$  V *vs.* RHE within the frequency range from 0.01 Hz to 1 000 000 Hz.

The electrochemically active surface area (ECSA) was determined *via* measuring the double-layer capacitance ( $C_{dl}$ ), which was derived from cyclic voltammetry (CV) curves obtained in the narrow potential range of 0.0–0.1 V (*vs.* Ag/AgCl) at scan rates of 2, 4, 6, 8, 10, 20, 40, 60, 80, and 100 mV s<sup>-1</sup>. 1 M KOH was used as the electrolyte and N<sub>2</sub> was used as the feed gas.

The adsorption of OH<sup>-</sup> as a surrogate of CO<sub>2</sub><sup>-</sup> was examined based on a cathodic LSV scan between  $-0.4$  V and 0.0 V (*vs.* Ag/AgCl) in N<sub>2</sub>-saturated aqueous 1 M KOH solution at a scan rate of 20 mV s<sup>-1</sup>.

### Characterization of materials

Transmission electron microscopy (TEM), selected area electron diffraction (SAED) high-resolution transmission electron microscopy (HRTEM), high-angle annular dark field scanning transmission electron microscopy (HAADF-STEM), and energy dispersive X-ray spectroscopy (EDS) were performed using FEG-TEM apparatus (JEM-2100F, JEOL, Japan) operated at 200 kV. Scanning electron microscopy (SEM) analysis was performed using a field-emission scanning electron microscope (FESEM) (Hitachi S-4800, Japan). Powder X-ray diffraction (PXRD) analysis was carried out with a D/MAX-2000 diffractometer (Rigaku, Japan) using Cu K $\alpha$  radiation. X-ray photoelectron spectroscopy (XPS) analysis was conducted using an Axi Ultra imaging photoelectron spectrometer (Kratos, UK). Inductively coupled plasma optical emission spectroscopy (ICP-OES) analysis was performed using a Profile Spec ICP-OES spectrometer (Leeman, USA). All GC spectrometry experiments were carried out and recorded using GC-2014 apparatus (Shimadzu, Japan). <sup>1</sup>H-NMR spectroscopy analysis was conducted using a Bruker-500 MHz NMR spectrometer (Bruker, USA). Nitrogen adsorption-desorption isotherms were obtained using Micromeritics ASAP 2020 nitrogen adsorption apparatus (USA). Pore-size distributions were determined *via* the Barrett-Joyner-Halenda (BJH) method using the adsorption data.

## Results and discussion

### Morphology and structural analysis of the nanoporous hollow Au/CuO–CuO catalyst

Au/Cu<sub>2</sub>O<sub>4</sub>/C was firstly synthesized using a one-pot method. A composite of carbon black and Cu(II) oxalate was first prepared *via* co-precipitation, and Au was then introduced onto the surface of Cu<sub>2</sub>O<sub>4</sub> through the reduction of excess C<sub>2</sub>O<sub>4</sub><sup>2-</sup> in solution (see the Experimental section for more details). The results of transmission electron microscopy (TEM) and scanning electron microscopy (SEM) analysis (Fig. S1a–d and S2†) confirm the nanoporous hollow structure of Au/Cu<sub>2</sub>O<sub>4</sub>/C with a diameter of 200–500 nm. Energy dispersive X-ray spectroscopy (EDS) line scan measurements (Fig. S3†) further confirm this hollow structure. Selected area electron diffraction (SAED), high angle annular dark field scanning TEM (HAADF-STEM), and EDS mapping images (Fig. S1e–k†) demonstrate that most of the nanoporous hollow particles in Au/Cu<sub>2</sub>O<sub>4</sub>/C were composed of Cu<sub>2</sub>O<sub>4</sub>, while some Cu<sub>2</sub>O<sub>4</sub> nanoparticles were covered with Au nanoparticles (Au/Cu<sub>2</sub>O<sub>4</sub> composite). The



powder X-ray diffraction (XRD) pattern (Fig. S4†) is highly consistent with that of a mixture of  $\text{Cu}_2\text{O}_4 \cdot x\text{H}_2\text{O}$  and Au. For comparison, we also obtained  $\text{Cu}_2\text{O}_4/\text{C}$  and  $\text{Au}/\text{C}$  through similar methods, except that the addition of the Au precursor was omitted in the preparation process of  $\text{Cu}_2\text{O}_4/\text{C}$  and the addition of the Cu precursor was omitted in the preparation process of  $\text{Au}/\text{C}$ . Related characterization of  $\text{Cu}_2\text{O}_4/\text{C}$  and  $\text{Au}/\text{C}$  (Fig. S5 and S6†) illustrates the formation of  $\text{Cu}_2\text{O}_4$  in  $\text{Cu}_2\text{O}_4/\text{C}$  and Au in  $\text{Au}/\text{C}$ . X-ray photoelectron spectroscopy (XPS) analysis of  $\text{Au}/\text{Cu}_2\text{O}_4/\text{C}$ ,  $\text{Cu}_2\text{O}_4/\text{C}$ , and  $\text{Au}/\text{C}$  (Fig. S7 and S8†) shows the existence of  $\text{Au}^0$  and  $\text{Cu}^{2+}$  in  $\text{Au}/\text{Cu}_2\text{O}_4/\text{C}$ ,  $\text{Cu}^{2+}$  in  $\text{Cu}_2\text{O}_4/\text{C}$ , and  $\text{Au}^0$  in  $\text{Au}/\text{C}$ .

Thermogravimetric analysis (TGA) results (Fig. S9†) display that  $\text{Cu}_2\text{O}_4$  in  $\text{Au}/\text{Cu}_2\text{O}_4/\text{C}$  decomposed at a temperature between 275 °C and 300 °C; thus, the pyrolysis temperature of  $\text{Au}/\text{Cu}_2\text{O}_4/\text{C}$  was set as 400 °C, and the obtained sample was named AuCu400. The results from TEM, SEM, SAED, HAADF-STEM, EDS mapping, EDS line scanning, and XRD analysis (Fig. 1 and S10a and b†) confirm that  $\text{Cu}_2\text{O}_4$  in  $\text{Au}/\text{Cu}_2\text{O}_4/\text{C}$  decomposed to small CuO nanoparticles (20–50 nm in diameter), while  $\text{Au}/\text{Cu}_2\text{O}_4$  decomposed to  $\text{Au}/\text{CuO}$  nanoparticles. The nanoporous hollow structure (formed *via* the accumulation of CuO or  $\text{Au}/\text{CuO}$  nanoparticles, 200–500 nm in diameter) was still retained after annealing in air.  $\text{N}_2$  adsorption-desorption isotherms and pore-size distributions were employed to analyze

the pore structure of AuCu400. The obvious hysteresis loop between the adsorption and desorption branches and the peaks at around 3 nm, 10 nm, and 30 nm in the size distribution curve demonstrate the existence of piled mesopores, which is in good agreement with TEM and SEM imaging (Fig. S10c and d†).<sup>42</sup> In addition, this conclusion can be verified based on the pore volume of AuCu400 being  $0.263 \text{ cm}^3 \text{ g}^{-1}$ , which is about 17 times that of  $\text{Au400} + \text{NPCuO}$  ( $0.015 \text{ cm}^3 \text{ g}^{-1}$ ).

In comparison, Cu400 and Au400 were also prepared through a similar annealing procedure based on  $\text{Cu}_2\text{O}_4/\text{C}$  and  $\text{Au}/\text{C}$ , respectively, and characterization shows that Cu400 was composed of CuO while Au400 was composed of Au (Fig. S11–S14†). XPS spectra of AuCu400, Cu400, and Au400 (Fig. S15 and S16†) also show the existence of  $\text{Au}^0$  and  $\text{Cu}^{2+}$  in AuCu400,  $\text{Cu}^{2+}$  in Cu400, and  $\text{Au}^0$  in Au400.

### Evaluation of the electrocatalytic activity and selectivity during the $\text{eCO}_2\text{RR}$

AuCu400 was ultrasonically mixed with Nafion ionomer solution in isopropanol, and the resulting suspension was sprayed onto carbon paper with a gas diffusion layer to prepare an AuCu400 electrode. In comparison, electrodes containing Cu400, Au400, and  $\text{Au}/\text{Cu}_2\text{O}_4/\text{C}$  were also prepared *via* the same method, and they are denoted as Cu400, Au400, and AuCu electrodes, respectively.  $\text{eCO}_2\text{RR}$  measurements were performed in a flow cell using  $\text{IrO}_2/\text{C}$  as the counter electrode and 1 M KOH aqueous solution as the electrolyte. Linear sweep voltammetry (LSV) was first conducted to investigate the electrochemical activities of the four samples. As seen in Fig. 2a, we find that at potentials more negative than  $-0.35 \text{ V}$ , current densities followed the order:  $\text{AuCu400} > \text{Cu400} > \text{Au400}$ ,

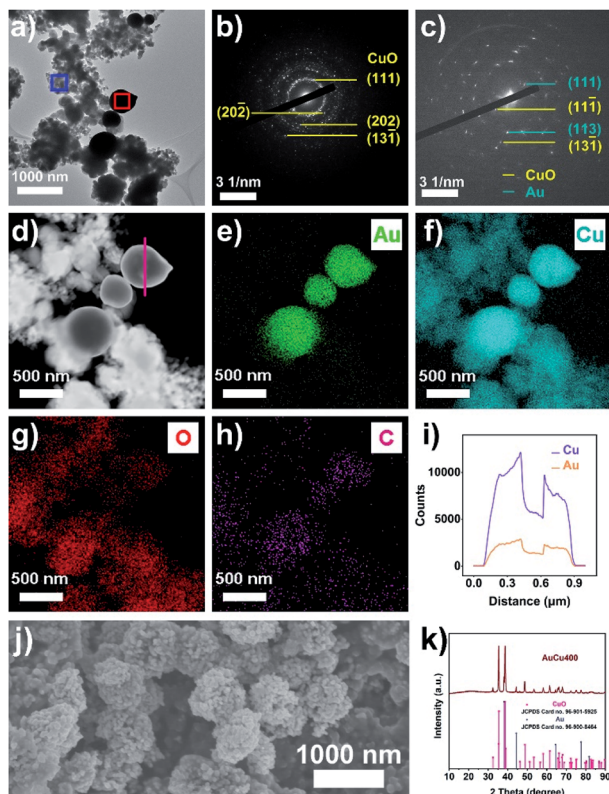


Fig. 1 (a) A TEM image of AuCu400; SAED patterns from the (b) blue square and (c) red square from (a); (d) an HAADF-STEM image and (e–h) EDS elemental mapping images of AuCu400; (i) the EDS line scanning result for AuCu400 (the scanning area is the pink line in (d)); (j) an SEM image of AuCu400; and (k) the XRD pattern of AuCu400.

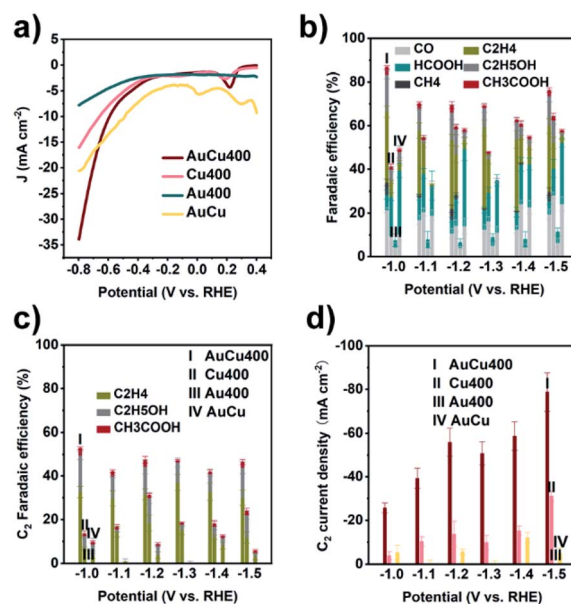


Fig. 2 (a) LSV curves, (b)  $\text{CO}_2$  reduction product FE values, (c)  $\text{C}_2$  product FE values, and (d)  $\text{C}_2$  product partial current densities from the four samples, using  $\text{CO}_2$  as the feed gas. In (b), "I", "II", "III", and "IV" refer to AuCu400, Cu400, Au400, and AuCu, respectively.



meaning that a combination of Au and Cu could ensure a larger current density. The electrochemically active surface areas (ECSAs) of AuCu, Cu400, and AuCu400 were measured based on CV curves (Fig. S17a–c†), and similar slopes were obtained (Fig. S17f†), indicating that the larger current density in the case of AuCu400 is not caused by an increase in the ECSA. In addition, at potentials more positive than  $-0.7$  V, AuCu exhibited the largest current density of the four samples, maybe resulting from the decomposition of  $\text{Cu}_2\text{O}_4$ .

The four samples were then examined as electrocatalysts for the  $\text{eCO}_2\text{RR}$  based on the controlled-potential electrolysis method. Fig. 2b and Table S1† provide the faradaic efficiency (FE) values for all products using the four electrodes. At all applied potentials, AuCu400 exhibited the highest  $\text{CO}_2$  reduction product selectivity and the lowest HER activity among the four catalysts. It could reach a maximum  $\text{CO}_2$  reduction product FE of 86.7% at  $-1.0$  V. Fig. 2c and d show the  $\text{C}_2$  product FE and  $\text{C}_2$  product partial current density values, respectively, of the four electrodes at different potentials, from which we could find that AuCu400 exhibited superior  $\text{C}_2$  product activity and selectivity. At  $-1.0$  V, the  $\text{C}_2$  product FE of AuCu400 reaches its maximum value of 52.8%, and the  $\text{C}_2$  product FE could remain above 40% over a wide potential range, from  $-1.0$  V to  $-1.5$  V. At  $-1.5$  V, AuCu400 exhibited the largest  $\text{C}_2$  product partial current density of  $78.77 \text{ mA cm}^{-2}$ . The catalytic performance of AuCu400 towards the  $\text{eCO}_2\text{RR}$  is compared with other values in the literatures, as listed in Table S2.† AuCu400 showed comparable  $\text{C}_2$  FE values to other Au–Cu-based tandem catalysts (Table S2-1†) and the widest potential window for  $\text{C}_2$ -dominated selectivity. In contrast, its counterparts Cu400 and AuCu could only reduce  $\text{CO}_2$  into  $\text{C}_2$  products with poor FEs ( $<20\%$ ) and small partial current densities ( $<40 \text{ mA cm}^{-2}$ ) at all applied potentials, while Au400 produced no  $\text{C}_2$  products at all. We note the poor stability of AuCu400 over long periods of electrolysis (Fig. S18†), which is due to commonly reported stability issues related to flooding in carbon-paper-based gas diffusion electrodes.<sup>43–46</sup> To rule out the possibility that C-based products were produced upon the decomposition of electrolyte ( $\text{CO}_2$  reacts with KOH to form  $\text{K}_2\text{CO}_3$  or  $\text{KHCO}_3$ ), electrolysis with AuCu400 using  $\text{N}_2$  as the feed gas was performed; the much lower current density and lack of detection of C-based products confirm that C-based products are produced from the reduction of  $\text{CO}_2$  (Fig. S19 and S20†). Significantly, although  $\text{N}_2$  LSV also showed a modest HER onset peak at the same potential, the  $\text{H}_2$  product FE of AuCu400 at  $-1.0$  V is only 4.8% (Table S1†). Therefore, it is reasonable to assume that AuCu400 would show a more preferential catalytic effect toward the  $\text{eCO}_2\text{RR}$  than the HER when using  $\text{CO}_2$  as the feed gas.

### The conditions required for $\text{eCO}_2\text{RR}$ tandem catalysis

In order to investigate the formation mechanism of  $\text{C}_2$  products over AuCu400,  $\text{eCORR}$  measurements were performed over AuCu400 and Cu400. A comparison of the results (Fig. 3a) demonstrates that AuCu400 exhibited no enhancement in  $\text{C}_2$  formation compared to Cu400 under  $\text{CO}$ -reducing conditions at three applied potentials, which illustrates that nanoporous

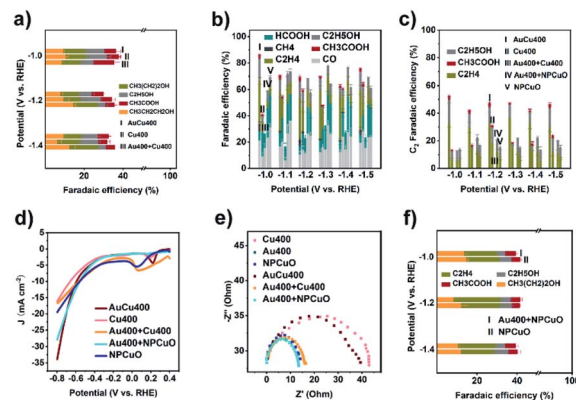


Fig. 3 (a)  $\text{C}_2$  product FEs of AuCu400, Cu400, and Au400 + Cu400 using  $\text{CO}$  as the feed gas; (b)  $\text{CO}_2$  reduction product FEs, (c)  $\text{C}_2$  product FEs; (d) LSV curves of AuCu400, Cu400, Au400 + Cu400, Au400 + NPCuO, and NPCuO using  $\text{CO}_2$  as the feed gas; (e) Nyquist plots for Cu400, Au400, AuCu400, Au400 + Cu400, Au400 + NPCuO, and NPCuO; and (f)  $\text{C}_2$  product FEs of Au400 + NPCuO and NPCuO using  $\text{CO}$  as the feed gas. In (b), “I”, “II”, “III”, “IV”, and “V” refer to AuCu400, Cu400, Au400 + Cu400, Au400 + NPCuO, and NPCuO, respectively.

hollow Au/CuO in AuCu400 did not influence CuO active sites and that only the CuO component was responsible for the  $\text{eCORR}$  in the cases of both AuCu400 and Cu400. To further illustrate the reaction process, we also examined the  $\text{CO}$  production rates (Fig. S21a†). As expected, AuCu400 has a greater  $\text{CO}$  reduction rate than Cu400, which means that the introduction of Au/CuO sites generates more  $\text{CO}$ . In addition, the higher ratio of  $\text{C}_2$  products to  $\text{C}_1$  products can imply that the enhancement is due to an increase in the rate of C–C coupling (Fig. S21b†).<sup>29</sup> Our experiments strongly support the conclusion that the nanoporous hollow Au/CuO and CuO active sites in AuCu400 operated independently. Au/CuO produced  $\text{CO}^*/\text{CO}$  from  $\text{CO}_2$ , and this generated  $\text{CO}^*/\text{CO}$ , in addition to  $\text{CO}^*/\text{CO}$  generated on CuO itself, was further utilized for multi-carbon production over the Cu surface (Fig. 4a).<sup>47</sup> Compared to bare CuO sites, nanoporous hollow Au/CuO is much more suitable for the formation of  $\text{CO}^*/\text{CO}$  due to its more optimal binding strength toward  $^*\text{COOH}$ , which endows AuCu400 with higher  $\text{C}_2$  product selectivity.

In some published literature reports, this kind of catalytic mechanism, which is known as “tandem catalysis”, has been reported. However, most of these research reports focus on the fabrication of tandem catalysts *via* electrochemical deposition methods or physical mixing processes, and they have not explored the conditions under which a tandem catalysis process will proceed most effectively and efficiently. To solve this problem, we prepared Au400 + Cu400 (Fig. S22†) *via* physically mixing the above-mentioned Au400 and Cu400 materials at the same Au/Cu mass ratio found in AuCu400 (determined *via* ICP-OES to be 1 : 5.59). Unexpectedly, the catalytic performance data from the  $\text{eCO}_2\text{RR}$  using AuCu400, Cu400, and Au400 + Cu400, shown in Fig. 3b–d and Table S1,† illustrates that Au400 + Cu400 only produced  $\text{CO}$ , formate, and  $\text{H}_2$  at all applied potentials, which is completely different from the other samples



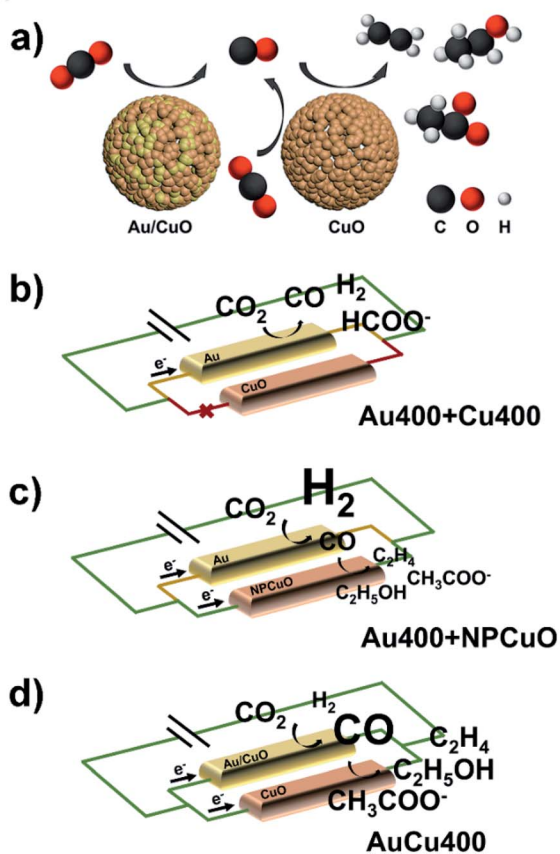


Fig. 4 (a) The formation mechanism of  $C_2$  products on AuCu400; and the proposed origin of the different  $C_2$  FE performances of (b) Au400 + Cu400, (c) Au400 + NPCuO, and (d) AuCu400. Only tandem catalysis processes are shown in (b)–(d).

containing Cu, and it also exhibited a relatively low current density. However, eCORR measurements of Au400 + Cu400 showed almost the same results as for AuCu400 and Cu400 (Fig. 3a), proving that CuO, the source of CO/\*CO-reduction sites in Au400 + Cu400, is still effective under CO-reducing conditions. But in the eCO<sub>2</sub>RR, Cu400, which is dominant in Au400 + Cu400, seemed to be “silenced” by the minor component Au400. Considering that the CuO sites in AuCu400 are not influenced by Au/CuO sites in the reaction process, we speculate that there are some undetected factors leading to the differences in reaction pathways, which are manifested in the failure of CuO sites in Au400 + Cu400 to be involved in the reaction where, in other words, tandem catalysis did not occur.

In order to explore these factors, electrochemical impedance spectroscopy (EIS) studies were then conducted on Au400, Cu400, AuCu400, and Au400 + Cu400 to analyze the electron transfer process. The obtained Nyquist plots (Fig. 3e) indicate that the interfacial charge-transfer resistance of Au400 was much lower than that of Cu400, and this may result from the high conductivity of Au and the low conductivity of CuO. In the catalytic process involving Au400 + Cu400, almost all of the electrons are transferred to Au sites in Au400 due to the huge difference in charge-transfer resistance between Au400 and Cu400; therefore, the CuO sites in Cu400 were “short circuited”

and failed to play any role in the reduction of CO<sub>2</sub> (Fig. 4b). The small difference in charge-transfer resistance between Au400 and Au400 + Cu400 also supports this reasoning. Although we failed to synthesize a pure Au/CuO phase (because once the proportion of Au was further increased, Au formed an independent phase in the form of larger Au nanoparticles instead of Au/CuO), we can also infer that the electrical conductivity of nanoporous hollow Au/CuO is not far apart from that of CuO because the EIS plot diameter of AuCu400 is close to that of Cu400. Therefore, as a necessary condition for the occurrence of tandem catalysis, the matching of charge-transfer resistance between CO/\*CO-production sites and CO/\*CO-reduction sites must be present to ensure the high  $C_2$  product selectivity of AuCu400.

To check this point, we then prepared Au400 + NPCuO *via* physically mixing Au400 and commercial CuO nanoparticles (NPCuO) at a Au/Cu ratio of 1 : 5.59. As a control sample for Cu400, NPCuO shows a similar size, XRD pattern, ECSA, and Cu 2p XPS spectrum to Cu400 (Fig. S17c and d and S23†). More importantly, as for Au400 + NPCuO, the small difference in charge-transfer resistance between Au400 and NPCuO (Fig. 3e) could ensure that the Au sites and CuO sites both gain enough electrons during the reduction process, which provides the possibility for the occurrence of tandem catalysis.

We then compared the catalytic performance towards the eCO<sub>2</sub>RR of AuCu400, Cu400, Au400 + NPCuO, and NPCuO. Au400 + NPCuO could convert CO<sub>2</sub> into  $C_2$  products at most applied potentials. At potentials more positive than  $-1.0$  V, the  $C_2$  product FEs of these four catalysts followed the order: AuCu400 > Cu400 > Au400 + NPCuO > NPCuO or AuCu400 > Au400 + NPCuO > Cu400 > NPCuO (Fig. 3b and c and Table S1†), which shows that the mixing of Au400 and NPCuO could enhance the selectivity for  $C_2$  products of NPCuO, while AuCu400 still exhibited the best  $C_2$  product FE. The ECSAs of Au400 + NPCuO and NPCuO were also similar (Fig. S17d–f†), indicating that the introduction of Au could also enhance the current density of NPCuO (Fig. 3d). In addition, it should be noted that the two mixed samples, Au400 + Cu400 and Au400 + NPCuO, show similar large current densities at potentials more positive than  $-0.7$  V, which may be due to the role of Au400 component. Combining this information with the eCORR performance test results of Au400 + NPCuO and NPCuO (Fig. 3f), we could point out that a tandem catalysis mechanism also exists in the eCO<sub>2</sub>RR catalytic process when using Au400 + NPCuO as the electrode, which could account for the enhancement of the  $C_2$  product FE of Au400 + NPCuO *versus* NPCuO. This further proves that the matching of charge-transfer resistance is a necessary condition for tandem catalysis.

The improvement of the  $C_2$  product selectivity of AuCu400 *versus* Cu400 and Au400 + NPCuO *versus* NPCuO seems to provide evidence that the introduction of CO/\*CO production sites to CuO sites would absolutely enhance the selectivity for  $C_2$  products. However, at all applied potentials, the  $C_2$  product FEs of Au400 + NPCuO were much lower than those of AuCu400, especially when focusing on the point that the extent of the  $C_2$  FE increase of Au400 + NPCuO compared with NPCuO is much lower than that of AuCu400 compared with Cu400; we attribute this to a mismatch of the optimal working potentials of Au sites



and CuO sites. Most research expounds that pristine Cu or CuO NPs promote C–C coupling most effectively at potentials more negative than  $-1.0$  V (this conclusion is also shown based on the catalytic performances of Cu400 and NPCuO towards the eCORR in this work, as shown in Fig. S24;† the  $C_2$  product FEs are significantly lower at  $-0.8$  V), but pristine Au NPs can only convert  $CO_2$  to CO at potentials in the small range of  $-0.6$  V to  $-0.8$  V; at potentials more negative than  $-1.0$  V, the formation of  $H_2$  is much more favorable on the surface of Au (this conclusion is also shown based on the catalytic performance of Au400 towards the e $CO_2$ RR in this work, as shown in Fig. S25;† the CO product FEs are significantly higher at potentials more negative than  $-1.0$  V). As a result, at potentials more negative than  $-1.0$  V, the amount of CO that Au could produce was relatively small, which restricted the further formation of  $C_2$  products at CuO sites (Fig. 4c). This may be the reason why the majority of tandem catalysts show high selectivity for  $C_2$  products only in a narrow potential window (about 0.2 V) near  $-1.0$  V.

To further verify this inference, we probed  $OH^-$  adsorption as a surrogate for  $CO_2^{*}$  adsorption through oxidative LSV scans in  $N_2$ -saturated 1 M KOH solution (Fig. S26†). Since the protonation of adsorbed  $CO_2^{*}$  is generally considered to be the rate-determining step in CO/\*CO production, the potential for surface  $OH^-$  adsorption can characterize the capabilities for stabilizing  $CO_2^{*}$  intermediates and producing CO/\*CO in the reaction path towards  $C_{2+}$  products.<sup>48–51</sup> The results reveal that AuCu400 shows the lowest potential, 37 mV, 51 mV, and 67 mV lower than NPCuO, Au400, and Cu400, respectively. Based on this information, we could infer that nanoporous hollow Au/CuO in AuCu400 might be capable of converting  $CO_2$  into CO at potentials more negative than  $-1.0$  V much more selectively and effectively than Au in Au400 (and in Au400 + NPCuO) because of changes to the intrinsic electronic properties, and the matching of the optimal working potentials of Au/CuO and CuO in AuCu400 ensured an effective and efficient tandem catalysis process, thus endowing AuCu400 with satisfactory  $C_2$  product activity and selectivity (Fig. 4d) over a wide potential range.

## Conclusions

In summary, we have developed a nanoporous hollow Au/Cu–CuO tandem catalyst, which exhibited a  $C_2$  product FE of 52.8% at  $-1.0$  V and a  $C_2$  product partial current density of 78.77 mA  $cm^{-2}$  at  $-1.5$  V. In addition, the  $C_2$  product FE remained stably above 40% over a wide potential range from  $-1.0$  V to  $-1.5$  V. The CO/\*CO production sites (Au/CuO) and CO/\*CO-reduction sites (CuO) showed good matching in terms of their optimal working potentials and charge-transfer resistance values, ensuring that a sufficient amount of CO was produced and there was sufficient electron supply to CuO sites at potentials more negative than  $-1.0$  V, thus further promoting the formation rate of  $C_2$  products to a satisfactory level. This work not only successfully fabricated a Au/CuO–CuO tandem nanocatalyst, but it also revealed the conditions under which CO/\*CO-production and CO/\*CO-reduction site pairing would cooperate most efficiently and under which the tandem catalysis process would proceed

most effectively; this may guide us in the development of new tandem e $CO_2$ RR catalysts with satisfactory performance.

## Conflicts of interest

There are no conflicts to declare.

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