

Cite this: *Chem. Sci.*, 2021, 12, 1272

All publication charges for this article have been paid for by the Royal Society of Chemistry

# On-surface isostructural transformation from a hydrogen-bonded network to a coordination network for tuning the pore size and guest recognition†

Dong-Dong Zhou,<sup>†a</sup> Jun Wang,<sup>‡b</sup> Pin Chen,<sup>‡c</sup> Yangyong He,<sup>b</sup> Jun-Xi Wu,<sup>a</sup> Sen Gao,<sup>c</sup> Zhihao Zhong,<sup>b</sup> Yunfei Du,<sup>c</sup> Dingyong Zhong<sup>\*b</sup> and Jie-Peng Zhang<sup>ID\*<sup>a</sup></sup>

Rational manipulation of supramolecular structures on surfaces is of great importance and challenging. We show that imidazole-based hydrogen-bonded networks on a metal surface can transform into an isostructural coordination network for facile tuning of the pore size and guest recognition behaviours. Deposition of triangular-shaped benzotrisimidazole (H<sub>3</sub>btim) molecules on Au(111)/Ag(111) surfaces gives honeycomb networks linked by double N–H···N hydrogen bonds. While the H<sub>3</sub>btim hydrogen-bonded networks on Au(111) evaporate above 453 K, those on Ag(111) transform into isostructural [Ag<sub>3</sub>(btim)] coordination networks based on double N–Ag–N bonds at 423 K, by virtue of the unconventional metal–acid replacement reaction (Ag reduces H<sup>+</sup>). The transformation expands the pore diameter of the honeycomb networks from 3.8 Å to 6.9 Å, giving remarkably different host–guest recognition behaviours for fullerene and ferrocene molecules based on the size compatibility mechanism.

Received 17th September 2020

Accepted 13th November 2020

DOI: 10.1039/d0sc05147k

rsc.li/chemical-science

## Introduction

Surface-supported supramolecular structures, self-assembled from deposited molecules and/or metal atoms/ions *via* van der Waals interactions, hydrogen bonds, coordination bonds, *etc.*, have received continuous interest in recent years due to their straightforward and unique (supramolecular) chemistry, as well as great potential in functional molecular devices,<sup>1–7</sup> although the synthesis difficulty, expense, scalability and applicability of these structures are still controversial. By applying heat, electric field, and/or light, or by further depositing other components, surface supramolecular structures may rearrange, and transform from hydrogen-bonded supramolecules into coordination ones, and even from supramolecules into macromolecules.<sup>8–13</sup>

Similar to three-dimensional (3D) solids, surface-supported 2D structures can have pores and serve as hosts for guest

recognition.<sup>14–21</sup> For example, deposition of linear dicarboxylic acid on a Cu surface could give hydrogen-bonded tetramers, and further deposition of Fe atoms could give ladder-type coordination networks with rectangular cavities, whose pore sizes can be tuned to accommodate one, two or three fullerene (C<sub>60</sub>) molecules, by using dicarboxylic acid with different lengths.<sup>22</sup> Some 3D crystalline materials can be post-synthetically modified to achieve the fine-tuning of structures and properties, by reacting with external chemicals or under physical stimuli such as heat and light.<sup>23–26</sup> Surface-supported 2D structures also have such reactivity. Note that the substrate can not only affect the surface structures but can also serve as the reactant.<sup>27–31</sup> However, the initial and final 2D structures generally differ a lot so it is difficult to predict or tailor the structures and properties.

The protons on the ligands may be replaced by metal atoms (protons were reduced by metal atoms to hydrogen), but the connection modes between hydrogen-bonding and coordination-bonding are generally different, making it difficult to predict 2D structures before and after transformation.<sup>28,32–34</sup> The linear coordination mode of monovalent coinage metal ions resembles those of strong hydrogen bonds, which has been well demonstrated in metal imidazolate frameworks.<sup>35–37</sup> However, surface-supported coordination networks based on imidazole derivatives have not been reported, probably because imidazole derivatives and their metal complexes can seldom form planar supramolecular structures. We designed benzo[1,2-*d*:3,4-*d'*:5,6-*d''*]trisimidazole (H<sub>3</sub>btim) as a rigid planar

<sup>a</sup>MOE Key Laboratory of Bioinorganic and Synthetic Chemistry, School of Chemistry, Sun Yat-Sen University, Guangzhou 510275, China. E-mail: zhangjp@mail.sysu.edu.cn

<sup>b</sup>School of Physics, State Key Laboratory of Optoelectronic Materials and Technologies, Sun Yat-Sen University, Guangzhou 510275, China. E-mail: dyzhong@mail.sysu.edu.cn

<sup>c</sup>National Supercomputer Center in Guangzhou, School of Data and Computer Science, Sun Yat-Sen University, Guangzhou 510006, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0sc05147k

‡ Authors share contribution.





Fig. 1 Topological transformation from a hydrogen-bonded network based on N–H···N bonds to an isostructural coordination network based on N–Ag–N bonds on Ag surfaces.

imidazole derivative suitable for forming 2D planar honeycomb-like networks connected by doubled N–H···N hydrogen bonds. Although in the bulk crystalline form the extremely small intralayer cavities of the staggered-stacking 2D honeycomb networks are inaccessible to any guest,<sup>38</sup> and H<sub>3</sub>btim in the solution/solid state is difficult to react with metals or metal ions to form coordination compounds, we show here that hydrogen-bonded H<sub>3</sub>btim honeycomb networks on Ag(111) can react with metal surfaces and transform into isostructural coordination networks (Fig. 1), which readily expands the pore size and drastically changes the guest recognition behaviors toward C<sub>60</sub> and ferrocene (Fe(Cp)<sub>2</sub>) molecules.<sup>39,40</sup>

## Results and discussion

### Honeycomb-like hydrogen-bonded networks

Under ultrahigh vacuum in scanning tunneling microscope (STM) equipment (*ca.* 10<sup>−10</sup> mbar), H<sub>3</sub>btim can evaporate at 503 K. For comparison, under common vacuum (*ca.* 10<sup>−4</sup> mbar) and ambient pressure, the evaporation temperature needs to be 623 K and 673 K, respectively.<sup>38</sup> STM showed that the H<sub>3</sub>btim molecules deposited on Au(111) and Ag(111) surfaces form large-area, regular and ordered honeycomb-like patterns resembling that of hydrogen-bonded layer fragments of its

crystal structure (Fig. 2 and S1†).<sup>38</sup> The lattice period (*T*) and the diameter of the hole (*d*) of the honeycomb-like patterns were measured to be ~12.0 Å and ~3.8 Å, respectively, consistent with the crystallographic results (12.7 Å and 3.8 Å, Fig. 2a and b). In the high-resolution STM image (Fig. 2c), H<sub>3</sub>btim molecules were present as rounded triangular protrusions individually, and there is a hollow inter-space between two adjacent molecules, corresponding to N–H···N hydrogen bonds (Fig. 1). These results demonstrated that the honeycomb-like structures indeed interconnect through double N–H···N hydrogen bonds. Although similar surface-supported hydrogen-bonded honeycomb-like 2D structures have been reported,<sup>41,42</sup> H<sub>3</sub>btim is the first example purely linked by N–H···N hydrogen bonds.

While H<sub>3</sub>btim molecules show the same self-assembled structure on Au and Ag surfaces, they possess distinctly different stabilities and reactivity. By gradually increasing the substrate temperature, the honeycomb-like structures on Au(111) disappeared at 453 K (Fig. S2 and S3†), which is lower than that of bulk H<sub>3</sub>btim, indicating that the interaction between H<sub>3</sub>btim molecules and the Au(111) surface is weaker than the π–π interaction in the bulk H<sub>3</sub>btim crystal. The physisorption feature is also implied by the unchanged herringbone reconstruction of the Au(111) surface underneath the self-assembled H<sub>3</sub>btim networks (Fig. S2 and S3†).

### Honeycomb-like coordination networks

For the Ag(111) surface, additional bright spots/rods appeared between the H<sub>3</sub>btim molecules, when the temperature was increased to 373 K, resulting in disordering of the honeycomb networks. The transformation completed at 423 K with the recurrence of ordered honeycomb networks (Fig. 3 and S4†). However, the measured *T* and *d* increased to ~14.0 Å and ~6.9 Å, due to the enlarged intermolecular spacing induced by intercalation of Ag atoms. To our knowledge, so far there has been no metal complex of H<sub>3</sub>btim or its deprotonated forms reported. We simulated the 2D honeycomb-like [Ag<sub>3</sub>(btim)] structure by the Molecular Mechanics (MM) method, giving a *T* of 14.4 Å and *d* of 6.8 Å, consistent with the measured values (Fig. 3). In the high-resolution STM image, the morphology of H<sub>3</sub>btim molecules transformed from a rounded triangle to



Fig. 2 (a) The 2D hydrogen-bonded network in H<sub>3</sub>btim bulk crystals. (b) The large-scale (*U* = −0.2 V, *I* = 1 nA) and (c) high-resolution (*U* = −0.6 V, *I* = 100 pA) STM image of H<sub>3</sub>btim molecules on Au(111) surfaces. The unit cell is indicated in black/white in (a)/(b). Molecular models are superimposed on the respective image.





Fig. 3 (a) MM simulated structure of  $[\text{Ag}_3(\text{btim})]$ . (b) The large-scale ( $U = 1.6 \text{ V}$ ,  $I = 100 \text{ pA}$ ) and (c) high-resolution ( $U = 0.08 \text{ V}$ ,  $I = 1.54 \text{ nA}$ ) STM image of  $[\text{Ag}_3(\text{btim})]$  on Ag(111) surfaces. The unit cell is indicated in black/white in (a)/(b). Molecular models are superimposed on the respective image.

a sharp one, corresponding to the redistribution of molecular orbitals by deprotonation. Between each pair of adjacent triangles, a fusiform spot appeared and can be assigned to the two Ag(I) ions connecting the  $\text{btim}^{3-}$  ligands (Fig. 3).<sup>43–45</sup>

It should be noted that the  $\text{H}_3\text{btim}$  hydrogen-bonded network can fully cover the Ag(111) surface, but partial desorption of  $\text{H}_3\text{btim}$  molecules from the surface occurred during the annealing process (Fig. S5<sup>†</sup>), and samples with an initial coverage varying from a submonolayer to monolayer could result in identical  $[\text{Ag}_3(\text{btim})]$  structures (Fig. S6<sup>†</sup>). The size of the  $[\text{Ag}_3(\text{btim})]$  networks on Ag(111) surfaces can be up to 200 nm (Fig. S5<sup>†</sup>), and an external metal source and continuous  $\text{H}_3\text{btim}$  molecule deposition would benefit the growth of larger  $[\text{Ag}_3(\text{btim})]$  networks.<sup>46</sup> Moreover, the reverse transformation from the coordination to hydrogen-bonded networks was not observed, being consistent with the irreversible hydrogen evolution process. Nevertheless, samples with both structures coexisting on the surface can be prepared by secondary deposition of  $\text{H}_3\text{btim}$  molecules on the surface with  $[\text{Ag}_3(\text{btim})]$  (Fig. S7<sup>†</sup>).

### Host-guest recognition

By using  $\text{C}_{60}$  or  $\text{Fe}(\text{Cp})_2$  with different sizes as the guest (Fig. 4a and S8<sup>†</sup>), the host-guest recognition behaviours of the isostructural hydrogen-bonded and coordination networks were investigated. STM showed that the large  $\text{C}_{60}$  molecules aggregated periodically with a hexagonal pattern on bare Ag(111) and on the hydrogen-bonded  $\text{H}_3\text{btim}$  network assembled on Ag(111) (Fig. S9<sup>†</sup>). The lattice period of  $\text{C}_{60}$  molecules was measured to be  $\sim 9.5 \text{ \AA}$ , being similar to the values ( $10.2 \text{ \AA}$ ) on other metal substrates,<sup>47</sup> and close to the molecular size ( $9.8 \text{ \AA}$ ), but obviously smaller than that of the  $\text{H}_3\text{btim}$  network on Ag(111) ( $12.7 \text{ \AA}$ ). On  $[\text{Ag}_3(\text{btim})]$ , low-concentration  $\text{C}_{60}$  molecules located randomly, but they showed very similar heights with a difference of less than  $0.2 \text{ \AA}$  (Fig. S10<sup>†</sup>), and the shortest intermolecular distances are  $\sim 14.0 \text{ \AA}$  (Fig. S10<sup>†</sup>), being consistent with the  $T$  value of  $[\text{Ag}_3(\text{btim})]$ , which indicates that they are located at the holes. For comparison,  $\text{C}_{60}$  molecules on  $\text{H}_3\text{btim}$  showed a large height difference of *ca.*  $0.5 \text{ \AA}$  (Fig. S9<sup>†</sup>), indicating that they locate at different environments. Nevertheless, high-concentration  $\text{C}_{60}$  molecules tend to aggregate on the Ag(111) surface (Fig. S11<sup>†</sup>).

On the  $\text{H}_3\text{btim}$  hydrogen-bonded networks,  $\text{Fe}(\text{Cp})_2$  molecules distributed randomly with adjacent distances of  $6.5\text{--}10.5 \text{ \AA}$  (Fig. 4b and S12<sup>†</sup>), far shorter than the  $T$  value of  $\text{H}_3\text{btim}$ . By contrast, on  $[\text{Ag}_3(\text{btim})]$ ,  $\text{Fe}(\text{Cp})_2$  molecules showed large-area regular patterns ( $T \sim 14.0 \text{ \AA}$ ) matching well with those of the holes of the host (Fig. 4c and d). These indicate that the larger holes of  $[\text{Ag}_3(\text{btim})]$ , compared with  $\text{H}_3\text{btim}$ , are suitable for accommodating  $\text{Fe}(\text{Cp})_2$ .

### Computational calculations

To explain the different host-guest recognition behaviours, the adsorption enthalpies ( $\Delta H$ ) of  $\text{C}_{60}$  and  $\text{Fe}(\text{Cp})_2$  at typical environments were calculated by density functional theory (DFT)

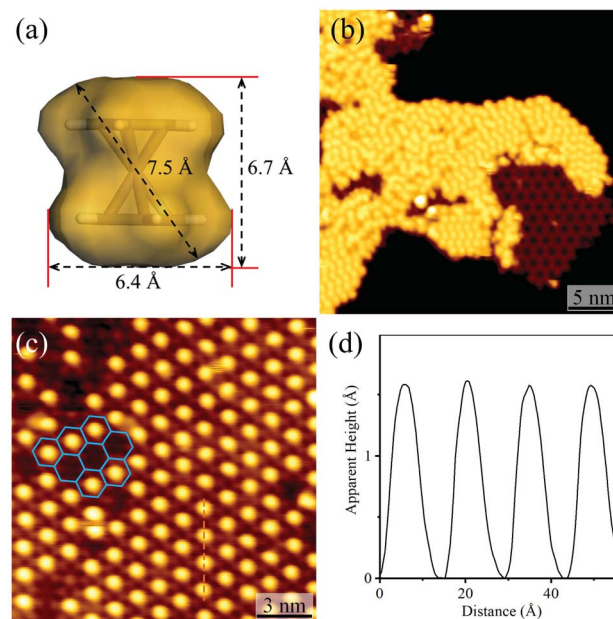


Fig. 4 Adsorption of  $\text{Fe}(\text{Cp})_2$  on  $\text{H}_3\text{btim}$  networks or  $[\text{Ag}_3(\text{btim})]$  networks. (a) The molecular size and shape of  $\text{Fe}(\text{Cp})_2$ . (b) STM image of  $\text{Fe}(\text{Cp})_2$  molecules on  $\text{H}_3\text{btim}$  networks ( $U = -3.0 \text{ V}$ ,  $I = 10 \text{ pA}$ ). (c) STM image of  $\text{Fe}(\text{Cp})_2$  molecules on  $[\text{Ag}_3(\text{btim})]$  networks ( $U = -2.0 \text{ V}$ ,  $I = 50 \text{ pA}$ ) with (d) the height profile along the orange line in Fig. 4c.



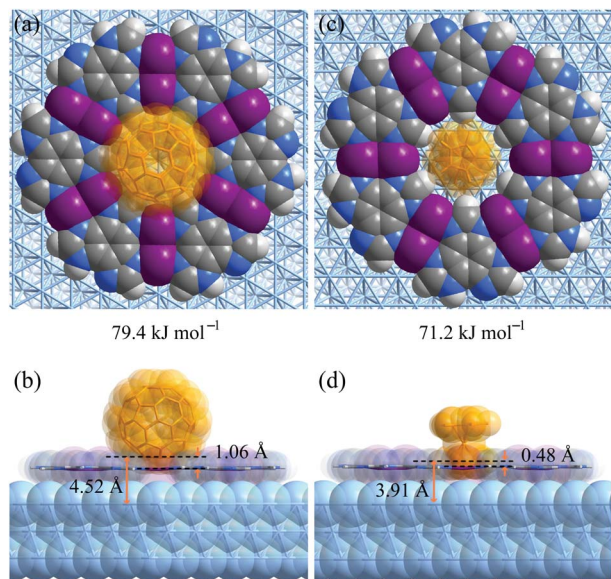


Fig. 5 DFT simulated primary host-guest structures on Ag(111) supported  $[Ag_3(btIm)]$  networks. (a and b)  $C_{60}$ . (c and d)  $Fe(Cp)_2$ . (a and c) Top views. (b and d) Side views.

simulations (Fig. 5 and S13–S17<sup>†</sup>). For  $C_{60}$  on  $H_3btIm$ ,  $\Delta H$  follows imidazole ring ( $-52.1 \text{ kJ mol}^{-1}$ ) < phenyl ring ( $-43.4 \text{ kJ mol}^{-1}$ ) < hole ( $-40.1 \text{ kJ mol}^{-1}$ ) < hydrogen bonds ( $-31.5 \text{ kJ mol}^{-1}$ ). For  $Fe(Cp)_2$  on  $H_3btIm$ ,  $\Delta H$  follows hydrogen bonds ( $-27.7 \text{ kJ mol}^{-1}$ ) < hole ( $-25.2 \text{ kJ mol}^{-1}$ ) < phenyl ring ( $-11.9 \text{ kJ mol}^{-1}$ ) < imidazole ring ( $-7.9 \text{ kJ mol}^{-1}$ ). Because  $C_{60}$  will be too crowded to fully occupy the primary adsorption site (imidazole ring), and  $Fe(Cp)_2$  exhibits little difference between the primary and secondary adsorption sites, these guests cannot form an ordered host-guest arrangement on  $H_3btIm$ . In contrast, the primary adsorption sites for  $C_{60}$  and  $Fe(Cp)_2$  on  $[Ag_3(btIm)]$  networks are both on the holes, and their  $\Delta H$  values ( $-79.4 \text{ kJ mol}^{-1}$  and  $-71.2 \text{ kJ mol}^{-1}$ ) are much lower than those of the secondary adsorption sites ( $-56.9 \text{ kJ mol}^{-1}$  for  $C_{60}$  on the phenyl ring and  $-42.8 \text{ kJ mol}^{-1}$  for  $Fe(Cp)_2$  on the double N–Ag–N bonds, respectively). Therefore,  $C_{60}$  and  $Fe(Cp)_2$  tend to locate on the holes of  $[Ag_3(btIm)]$  networks.

Comparison of the host-guest structures on the holes clearly explained the differences of recognition behaviors based on the size mechanism. On  $[Ag_3(btIm)]$ , the distances from the deepest inserting atom of  $C_{60}$  to the coordination network plane and the Ag(111) surface are 1.06 Å and 4.52 Å (Fig. 5a and b), which are significantly shorter than and slightly longer than the sum of the van der Waals radii of the corresponding atoms (C 1.7 Å, Ag 2.2 Å), *i.e.*, 3.4 Å and 3.9 Å, respectively. This means  $C_{60}$  can partially insert into the hole and cannot touch the Ag(111) surface. By contrast, these distances are 0.48 Å and 3.91 Å for  $Fe(Cp)_2$  on  $[Ag_3(btIm)]$ , indicating that the smaller  $Fe(Cp)_2$  molecule can fully insert to touch the Ag(111) surface (Fig. 5c and d). Therefore, the  $\Delta H$  difference of  $Fe(Cp)_2$  and  $C_{60}$  on  $[Ag_3(btIm)]@Ag(111)$  is far less than those of their molecular sizes and boiling points. For comparison, these distances for  $C_{60}/Fe(Cp)_2$  on  $H_3btIm@Ag(111)$  are 2.21/2.14 Å and 5.66/5.61 Å, respectively (Fig. S13<sup>†</sup>), consistent with relatively high  $\Delta H$  values.

The adsorption of  $C_{60}$  ( $-75.2 \text{ kJ mol}^{-1}$ ) on Ag(111) is slightly weaker than on the hole of  $[Ag_3(btIm)]$  (Fig. S18<sup>†</sup>). On Ag(111),  $C_{60}$  molecules can approach each other to furnish close and strong intermolecular interactions ( $-13.4 \text{ kJ mol}^{-1}$  with a center-center distance of 9.9 Å, Fig. S19<sup>†</sup>). On the holes of  $[Ag_3(btIm)]$ , intermolecular interactions between adjacent  $C_{60}$  molecules are much weaker ( $-3.9 \text{ kJ mol}^{-1}$  with a center-center distance of 14.4 Å, Fig. S19<sup>†</sup>). Therefore, two or more  $C_{60}$  molecules tend to aggregate on Ag(111) rather than periodically locate on the holes of  $[Ag_3(btIm)]$ . For comparison, the adsorption of  $Fe(Cp)_2$  on Ag(111) ( $-21.5 \text{ kJ mol}^{-1}$ , Fig. S18<sup>†</sup>) is much weaker than on the holes of  $[Ag_3(btIm)]$  ( $-71.2 \text{ kJ mol}^{-1}$ ).

## Conclusions

In summary, we investigated the self-assembly and reactivity of a high-symmetry imidazole derivative on Au(111)/Ag(111) surfaces, and achieved *in situ* transformation from a hydrogen-bonded network to an isostructural coordination network and fine tuning of pore sizes. Moreover, we directly observed the vastly different adsorption and recognition behaviors of guest molecules on these 2D porous networks, demonstrating the important role of pore size tailoring. These results can provide a new clue for the preparation and property modulation of new 2D materials, which have potential for energy conversion and storage.

## Experimental

Fullerene ( $C_{60}$ ) and ferrocene ( $Fe(Cp)_2$ ) were brought from Alfa Aesar (purity: 99.5%), and benzo[1,2-*d*:3,4-*d'*:5,6-*d''*]trisimidazole ( $H_3btIm$ ) was synthesized according to the literature,<sup>38</sup> and they were used without further purification.

STM measurements were carried out on an Omicron low-temperature STM system with a base pressure below  $1 \times 10^{-10}$  mbar. Clean metal substrates were obtained by repeated Ar<sup>+</sup> sputtering (at 298 K for 12 min) and annealing (at 750 K for 30 min). All the STM images were taken in the constant-current mode by using electrochemically etched tungsten tips with the samples cooled down with liquid nitrogen to 78 K. All given voltages refer to the bias on samples with respect to the STM tip.

A quartz crucible containing  $H_3btIm$  microcrystalline powders was put in the sample injection chamber. The chamber was degassed to *ca.*  $10^{-8}$  mbar and the crucible was heated to 473 K for several hours to remove impurities in  $H_3btIm$ . Subsequently, the crucible was heated at 503 K, and the sublimated  $H_3btIm$  molecules travelled through the molecular beam epitaxy (MBE) system to deposit onto the substrate surface (at 298 K). The obtained sample was then transferred from the preparation chamber to the STM chamber without exposure to air. The method for deposition of  $C_{60}$  molecules on the  $H_3btIm$  honeycomb network was the same as for deposition of  $H_3btIm$ , except that the sublimation/crucible temperature was 623 K and the deposition/substrate temperature was 273 K. Due to the easy evaporation/sublimation,  $Fe(Cp)_2$  powders were stored in a tin-foil crucible (diameter: 1.5 mm) rather than the quartz crucible. To start depositing  $Fe(Cp)_2$  molecules, a home-made tin-foil



crucible was transferred to the preparation chamber, and then put at a place near the target substrate (<200 K).

Molecular mechanics (MM) simulations were performed to obtain simulated a honeycomb-like [Ag<sub>3</sub>(btim)] structure by using the Forcite module in the Materials Studio 5.5 package. The initial configurations were produced by replacing H atoms of monolayer H<sub>3</sub>tim with Ag atoms. Structural optimization was based on the universal forcefield (UFF); all atoms and cell parameters were regarded as variable and  $Q_{eq}$  partial charges were employed. The cutoff radius was chosen as 18.5 Å for the LJ potential. Density functional theory (DFT) calculations were performed to obtain the adsorption enthalpies between hosts and guests by using the Vienna Ab-initio Simulation Package (VASP) package.<sup>48,49</sup> The Perdew–Burke–Ernzerhof (PBE) form of the generalized-gradient approximation (GGA) was used to treat the electronic exchange correlation. The energy cutoff was chosen to be 520 eV and the system was relaxed in the self-consistency accuracy of 10<sup>-4</sup> eV. All atoms in the unit cell were minimized by the conjugate gradient method until the force on each atom was less than 0.01 eV Å<sup>-1</sup>. The  $\Gamma$  point was used to integrate the Brillouin zone by the gamma-centered sampling method.

Since the H<sub>3</sub>btim structure is incommensurate with the Ag(111) surface, an extremely large supercell is required in order to match both of them, making it hard to handle during DFT calculations. For better comparison of the binding energy differences between the H<sub>3</sub>btim and [Ag<sub>3</sub>(btim)] host–guest systems, non-periodic networks on the Ag(111) surface were used. When considering the effect of the Ag(111) surface below the holes, a 7 × 7 × 7 Ag(111) supercell was used to construct the surface slab and a hexamer of H<sub>3</sub>btim or the corresponding [Ag<sub>3</sub>(btim)] fragment was built above the Ag(111) surface. Moreover, the Ag(111) surface slab built with 3 layers of Ag atoms and a vacuum space of 25 Å was employed to avoid interactions between the top and bottom surfaces. The topmost 2 layers were relaxed during optimization, while the remaining 1 layer was kept fixed to mimic the bulk.<sup>50–52</sup> The vdW-DF2 method<sup>53</sup> was employed to evaluate the van der Waals (vdW) effect in all calculations. Three potential binding sites (hydrogen bonds, the phenyl group, and the imidazole group of H<sub>3</sub>btim, while double N–Ag–N bonds, the phenyl group, and the imidazolate group of [Ag<sub>3</sub>(btim)]) were selected to calculate the adsorption enthalpies. C<sub>60</sub> and Fe(Cp)<sub>2</sub> were initially placed 2.5–3.0 Å above the hosts, and then the whole host–guest systems were optimized to the lowest energy position. The adsorption enthalpy ( $\Delta H$ ) is defined as follows:

$$\Delta H = E_{\text{host+guest}} - E_{\text{host}} - E_{\text{guest}}$$

where  $E_{\text{host+guest}}$ ,  $E_{\text{host}}$ , and  $E_{\text{guest}}$  are the energy of the final host–guest system, the pristine host, and the guest before adsorption in a vacuum environment, respectively.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work was supported by NSFC (21731007, 21821003, 11974431, and 21701191), the Guangdong Pearl River Talents Program (2017BT01C161), the Fundamental Research Funds for the Central Universities (18lgy42), and the Guangdong Province Key Area R&D Program (2019B010940001).

## Notes and references

- R. Gutzler, S. Stepanow, D. Grumelli, M. Lingenfelder and K. Kern, *Acc. Chem. Res.*, 2015, **48**, 2132–2139.
- S. De Feyter, A. Gesquière, M. M. Abdel-Mottaleb, P. C. M. Grim, F. C. De Schryver, C. Meiners, M. Sieffert, S. Valiyaveetil and K. Müllen, *Acc. Chem. Res.*, 2000, **33**, 520–531.
- G. B. Pan, J. M. Liu, H. M. Zhang, L. J. Wan, Q. Y. Zheng and C. L. Bai, *Angew. Chem., Int. Ed.*, 2003, **42**, 2747–2751.
- T. Chen, D. Wang and L. J. Wan, *Natl. Sci. Rev.*, 2015, **2**, 205–216.
- Q. Zeng and C. Wang, *Sci. China: Chem.*, 2010, **53**, 310–317.
- N. Crivillers, S. Furukawa, A. Minoia, A. Ver Heyen, M. Mas-Torrent, C. Sporer, M. Linares, A. Volodin, C. Van Haesendonck, M. Van der Auweraer, R. Lazzaroni, S. De Feyter, J. Veciana and C. Rovira, *J. Am. Chem. Soc.*, 2009, **131**, 6246–6252.
- Z. Zhang, Y. Li, B. Song, Y. Zhang, X. Jiang, M. Wang, R. Trumbleson, C. Liu, P. Wang, X. Q. Hao, T. Rojas, A. T. Ngo, J. L. Sessler, G. R. Newkome, S. W. Hla and X. Li, *Nat. Chem.*, 2020, **12**, 468–474.
- H. Liang, Y. He, Y. Ye, X. Xu, F. Cheng, W. Sun, X. Shao, Y. Wang, J. Li and K. Wu, *Coord. Chem. Rev.*, 2009, **253**, 2959–2979.
- L. Dong, P. N. Liu and N. Lin, *Acc. Chem. Res.*, 2015, **48**, 2765–2774.
- Q. Fan, J. M. Gottfried and J. Zhu, *Acc. Chem. Res.*, 2015, **48**, 2484–2494.
- J. A. A. W. Elemans, S. Lei and S. De Feyter, *Angew. Chem., Int. Ed.*, 2009, **48**, 7298–7332.
- R. Makiura, S. Motoyama, Y. Umemura, H. Yamanaka, O. Sakata and H. Kitagawa, *Nat. Mater.*, 2010, **9**, 565–571.
- C. Chen, P. Ding, M. Mura, Y. Chen, Y. Sun, L. N. Kantorovich, H. Gersen, F. Besenbacher and M. Yu, *Angew. Chem., Int. Ed.*, 2018, **57**, 16015–16019.
- J. A. Theobald, N. S. Oxtoby, M. A. Phillips, N. R. Champness and P. H. Beton, *Nature*, 2003, **424**, 1029–1031.
- M. O. Blunt, J. C. Russell, N. R. Champness and P. H. Beton, *Chem. Commun.*, 2010, **46**, 7157–7159.
- A. Kumar, K. Banerjee and P. Liljeroth, *Nanotechnology*, 2017, **28**, 082001.
- R. Sakamoto, K. Takada, T. Pal, H. Maeda, T. Kambe and H. Nishihara, *Chem. Commun.*, 2017, **53**, 5781–5801.
- M. Zhao, Y. Huang, Y. Peng, Z. Huang, Q. Ma and H. Zhang, *Chem. Soc. Rev.*, 2018, **47**, 6267–6295.
- M. S. Yao, J. J. Zheng, A. Q. Wu, G. Xu, S. S. Nagarkar, G. Zhang, M. Tsujimoto, S. Sakaki, S. Horike, K. Otake and S. Kitagawa, *Angew. Chem., Int. Ed.*, 2020, **59**, 172–176.



- 20 X. Peng, Y. Geng, M. Zhang, F. Cheng, L. Cheng, K. Deng and Q. Zeng, *Nano Res.*, 2018, **12**, 537–542.
- 21 J. Liu and C. Wöll, *Chem. Soc. Rev.*, 2017, **46**, 5730–5770.
- 22 S. Stepanow, M. Lingensfelder, A. Dmitriev, H. Spillmann, E. Delvigne, N. Lin, X. Deng, C. Cai, J. V. Barth and K. Kern, *Nat. Mater.*, 2004, **3**, 229–233.
- 23 S. M. Cohen, *Chem. Rev.*, 2011, **112**, 970–1000.
- 24 P. Deria, J. E. Mondloch, O. Karagiari, W. Bury, J. T. Hupp and O. K. Farha, *Chem. Soc. Rev.*, 2014, **43**, 5896–5912.
- 25 A. M. Wright, A. J. Rieth, S. Yang, E. N. Wang and M. Dincă, *Chem. Sci.*, 2018, **9**, 3856–3859.
- 26 Y. Ding, Y. P. Chen, X. Zhang, L. Chen, Z. Dong, H. L. Jiang, H. Xu and H. C. Zhou, *J. Am. Chem. Soc.*, 2017, **139**, 9136–9139.
- 27 D. Nieckarz and P. Szabelski, *J. Phys. Chem. C*, 2013, **117**, 11229–11241.
- 28 H. Walch, J. Dienstmaier, G. Eder, R. Gutzler, S. Schlogl, T. Sirtl, K. Das, M. Schmittel and M. Lackinger, *J. Am. Chem. Soc.*, 2011, **133**, 7909–7915.
- 29 L. Xing, Z. Peng, W. Li and K. Wu, *Acc. Chem. Res.*, 2019, **52**, 1048–1058.
- 30 A. Shchyrba, C. Wackerlin, J. Nowakowski, S. Nowakowska, J. Bjork, S. Fatayer, J. Girovsky, T. Nijs, S. C. Martens, A. Kleibert, M. Stohr, N. Ballav, T. A. Jung and L. H. Gade, *J. Am. Chem. Soc.*, 2014, **136**, 9355–9363.
- 31 P. Falcaro, K. Okada, T. Hara, K. Ikigaki, Y. Tokudome, A. W. Thornton, A. J. Hill, T. Williams, C. Doonan and M. Takahashi, *Nat. Mater.*, 2017, **16**, 342–348.
- 32 M. Lischka, R. Dong, M. Wang, N. Martsinovich, M. Fritton, L. Grossmann, W. M. Heckl, X. Feng and M. Lackinger, *Chem.–Eur. J.*, 2019, **25**, 1975–1983.
- 33 Q. N. Zheng, X. H. Liu, T. Chen, H. J. Yan, T. Cook, D. Wang, P. J. Stang and L. J. Wan, *J. Am. Chem. Soc.*, 2015, **137**, 6128–6131.
- 34 J. I. Urgel, D. Eciija, G. Lyu, R. Zhang, C. A. Palma, W. Auwarter, N. Lin and J. V. Barth, *Nat. Chem.*, 2016, **8**, 657–662.
- 35 J.-P. Zhang, Y.-B. Zhang, J.-B. Lin and X.-M. Chen, *Chem. Rev.*, 2012, **112**, 1001–1033.
- 36 K. S. Park, Z. Ni, A. P. Cote, J. Y. Choi, R. Huang, F. J. Uribe-Romo, H. K. Chae, M. O’Keeffe and O. M. Yaghi, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 10186–10191.
- 37 D.-D. Zhou, P. Chen, C. Wang, S.-S. Wang, Y. Du, H. Yan, Z.-M. Ye, C.-T. He, R.-K. Huang, Z.-W. Mo, N.-Y. Huang and J.-P. Zhang, *Nat. Mater.*, 2019, **18**, 994–998.
- 38 D.-D. Zhou, Y.-T. Xu, R.-B. Lin, Z.-W. Mo, W.-X. Zhang and J.-P. Zhang, *Chem. Commun.*, 2016, **52**, 4991–4994.
- 39 D. Zhong, T. Blömker, K. Wedeking, L. Chi, G. Erker and H. Fuchs, *Nano Lett.*, 2009, **9**, 4387–4391.
- 40 K. Wedeking, Z. Mu, G. Kehr, J. C. Sierra, C. M. Lichtenfeld, S. Grimme, G. Erker, R. Frohlich, L. Chi, W. Wang, D. Zhong and H. Fuchs, *Chem.–Eur. J.*, 2006, **12**, 1618–1628.
- 41 R. B. Lin, Y. He, P. Li, H. Wang, W. Zhou and B. Chen, *Chem. Soc. Rev.*, 2019, **48**, 1362–1389.
- 42 I. Hisaki, C. Xin, K. Takahashi and T. Nakamura, *Angew. Chem., Int. Ed.*, 2019, **58**, 11160–11170.
- 43 X. Zhang, N. Li, H. Wang, C. Yuan, G. Gu, Y. Zhang, D. Nieckarz, P. Szabelski, S. Hou, B. K. Teo and Y. Wang, *ACS Nano*, 2017, **11**, 8511–8518.
- 44 S. Zint, D. Ebeling, T. Schlöder, S. Ahles, D. Mollenhauer, H. A. Wegner and A. Schirmeisen, *ACS Nano*, 2017, **11**, 4183–4190.
- 45 F. Queck, O. Krejci, P. Scheuerer, F. Bolland, M. Otyepka, P. Jelinek and J. Repp, *J. Am. Chem. Soc.*, 2018, **140**, 12884–12889.
- 46 D. Kühne, F. Klappenberger, R. Decker, U. Schlickum, H. Brune, S. Klyatskaya, M. Ruben and J. V. Barth, *J. Am. Chem. Soc.*, 2009, **131**, 3881–3883.
- 47 T. Sakurai, X. D. Wang, T. Hashizume, V. Yurov, H. Shinohara and H. W. Pickering, *Appl. Surf. Sci.*, 1995, **87–88**, 405–413.
- 48 G. Kresse and J. Furthmüller, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 49 G. Kresse and J. Hafner, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **49**, 14251–14269.
- 50 D. Zhong, J.-H. Franke, S. K. Podiyanchari, T. Blömker, H. Zhang, G. Kehr, G. Erker, H. Fuchs and L. Chi, *Science*, 2011, **334**, 213–216.
- 51 G. Pawin, K. L. Wong, D. Kim, D. Sun, L. Bartels, S. Hong, T. S. Rahman, R. Carp and M. Marsella, *Angew. Chem., Int. Ed.*, 2008, **47**, 8442–8445.
- 52 D. Cui, M. Ebrahimi, F. Rosei and J. M. Macleod, *J. Am. Chem. Soc.*, 2017, **139**, 16732–16740.
- 53 J. Klimes, D. R. Bowler and A. Michaelides, *J. Phys.: Condens. Matter*, 2010, **22**, 022201.

