


 Cite this: *Phys. Chem. Chem. Phys.*,
2022, 24, 24002

 Received 18th June 2022,
Accepted 7th September 2022

DOI: 10.1039/d2cp02774g

rsc.li/pccp

Effect of conformational disorder on exciton states of an azobenzene aggregate†

 Evgenii Titov 

Azobenzene is a prototypical molecular photoswitch, widely used to trigger a variety of transformations at different length scales. In systems like self-assembled monolayers or micelles, azobenzene chromophores may interact with each other, which gives rise to the emergence of exciton states. Here, using first-principles calculations, we investigate how conformational disorder (induced, e.g., by thermal fluctuations) affects localization of these states, on an example of an H-type azobenzene tetramer. We find that conformational disorder leads to (partial) exciton localization on a single-geometry level, whereas ensemble-averaging results in a delocalized picture. The $\pi\pi^*$ and $n\pi^*$ excitons at high and low temperatures are discussed.

Aggregation of molecular chromophores gives rise to molecular excitons, featuring a collective nature of electronic excitation as opposed to excitation of individual molecules.^{1–3} The excitons play a crucial role in processes of light energy conversion.^{4,5} A particularly interesting class of light-triggered systems is photoswitchable materials.⁶ When photoswitch molecules are brought in close proximity to each other, exciton states are formed, as has been observed for azobenzene-containing self-assembled monolayers (SAMs)⁷ and micelles.⁸ The exciton formation has been proposed to hinder azobenzene isomerization,⁷ in addition to steric effects.^{9–11}

Exciton states arise as delocalized states in classical theories of exciton formation in molecular crystals¹ and in molecular dimers.² Yet, conformational disorder, induced, e.g., by thermal fluctuations, has been shown to lead to (partial) localization of the exciton states in polymers and oligomers.^{12–17} To the best of our knowledge, this effect has not been explored for systems of aggregated photoswitchable molecules.

In this work, using first-principles calculations, we study how conformational disorder caused by nuclear motion affects

the exciton states of a model azobenzene aggregate composed of four molecules (Fig. 1a). To this aim, we (i) perform ground-state Born–Oppenheimer molecular dynamics (BOMD) simulations employing dispersion-corrected density functional theory (DFT), (ii) calculate the electronic spectra for selected MD snapshots using linear response, long-range corrected time-dependent DFT (TD-DFT), and (iii) reveal the nature of the exciton states by means of a transition density matrix (TDM) analysis.

Specifically, in step (i) we perform B3LYP^{18,19}+D3(BJ)²⁰ constant-temperature, ground-state dynamics at ~ 300 K,[‡] for 10 ps. After that, 100 snapshots are selected from the BOMD trajectory (with a time interval of 100 fs) and subjected to TD-DFT calculations using the ω B97X-D functional²² (step ii). We use a long-range corrected functional in TD-DFT calculations to avoid a spurious charge transfer problem,²³ which has been observed for azobenzene dimers when using standard hybrid functionals like B3LYP.^{24,25} All calculations are performed with the def2-SV(P) basis set²⁶ using Gaussian 16.²⁷ Further details about the model and the methods are provided in SI1 (ESI†).

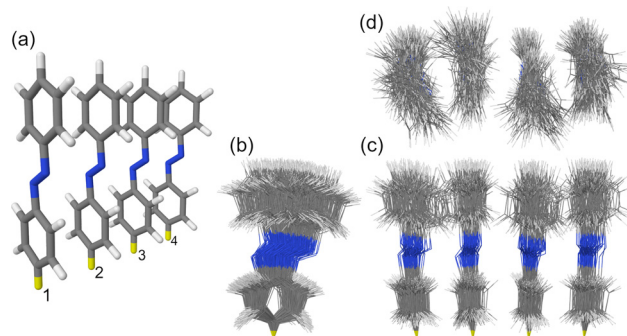


Fig. 1 (a) The H-type *trans* azobenzene tetramer studied in this work. The intermolecular separation distance is 3.5 Å. The yellow hydrogen atoms were fixed during MD simulations. The numbering of the monomers is shown at the bottom. (b) and (c) Side views and (d) top view of the ensemble of 100 snapshots selected from the MD trajectory at $T \approx 300$ K.

Theoretical Chemistry, Institute of Chemistry, University of Potsdam, Karl-Liebknecht-Straße 24-25, 14476 Potsdam, Germany. E-mail: titov@uni-potsdam.de

† Electronic supplementary information (ESI) available: Further details on models and methods, and further results. See DOI: <https://doi.org/10.1039/d2cp02774g>



In step (iii) we quantify localization and charge-transfer character of excited states using transition density matrix analysis.^{28,29} To this aim, the Löwdin-type^{30,31} “fraction of TDM” (FTDM) matrix \mathbf{F} (defined for a certain excited state at a given geometry) is computed:²⁴

$$F_{XY} = \frac{\sum_{\mu \in X} \sum_{\nu \in Y} (\mathbf{S}^{1/2} \mathbf{P}^{[\text{AO}]} \mathbf{S}^{1/2})_{\mu\nu}^2}{\sum_{\mu \in \text{complex}} \sum_{\nu \in \text{complex}} (\mathbf{S}^{1/2} \mathbf{P}^{[\text{AO}]} \mathbf{S}^{1/2})_{\mu\nu}^2}. \quad (1)$$

Here, $\mathbf{P}^{[\text{AO}]}$ is the TDM in atomic orbital (AO) basis (computed with Multiwfn 3.8³²) and \mathbf{S} is the AO overlap matrix. Diagonal elements F_{XX} quantify the contribution of local excitations and off-diagonal elements F_{XY} , $Y \neq X$ charge-transfer excitations (X , Y denote monomers of an aggregate). In what follows, we will often express the FTDM elements in %, *i.e.* as $F_{XY} \times 100\%$.

For an ensemble of N_s MD snapshots (labeled with α), we

compute the averaged FTDM matrix $\mathbf{F} = \frac{\sum_{\alpha} \mathbf{F}_{\alpha}}{N_s}$. Furthermore, for each snapshot α , we sort the diagonal of the \mathbf{F}_{α} matrix to introduce the highest (H), lowest (L), and intermediate ($H-1$ and $L+1$) monomers: $F_{\alpha,H} > F_{\alpha,H-1} > F_{\alpha,L+1} > F_{\alpha,L}$.³³ After that, we average these values over all snapshots, *e.g.*, for the H fragment $F_H = \frac{\sum_{\alpha} F_{\alpha,H}}{N_s}$. This procedure allows one to judge about exciton localization on the single-geometry level.^{17,33}

The studied H-type *trans* azobenzene tetramer is shown in Fig. 1a. The H-type arrangement is motivated by azobenzene packing in self-assembled monolayers.⁷ The used model with intermolecular distance of 3.5 Å is an extension of the densely-packed dimer model used in our previous works.^{10,34} The ensemble of 100 MD snapshots demonstrating the extent of conformational disorder at ~ 300 K is presented in Fig. 1b–d.

We first examine the $\pi\pi^*$ absorption band, corresponding to the bright, $S_0 \rightarrow S_2$ transition of the monomer. The spectra of the monomer and the tetramer calculated at the single geometries (with the monomer geometry optimized at the B3LYP/def2-SV(P) level) are shown in Fig. 2a. As can be expected for an H-aggregate, we observe a blue shift in comparison with the monomer spectrum, ~ 0.24 eV in our case. The brightest transition in the tetramer is the $S_0 \rightarrow S_{13}$ transition. [We note that the tetramer stick spectrum turned out to be rather sensitive to the monomer geometry—see spectra of tetramers constructed from the B3LYP/def2-SV(P) and B3LYP/def2-TZVP²⁶ optimized monomer geometries in Fig. S1 (ESI[†]) and FTDM matrices in Fig. S2, ESI[†].] Using the simplest exciton model (which includes only local excitations and assumes large energy gaps between monomeric states), one would expect the brightest transition to be $S_0 \rightarrow S_8$. However, at the small intermolecular separation (3.5 Å) and with the ω B97X-D functional, we observe that transitions from $S_0 \rightarrow S_8$ to $S_0 \rightarrow S_{11}$ originate from the $S_0 \rightarrow S_{3/4}$ monomeric transition, and the $S_0 \rightarrow S_{12}$ transition is a charge-resonance²⁹ transition corresponding to the monomeric $S_0 \rightarrow S_2$ transition. We also note that the $S_0 \rightarrow S_6$ transition in the tetramer at 3.47 eV has a small but nonzero oscillator strength of

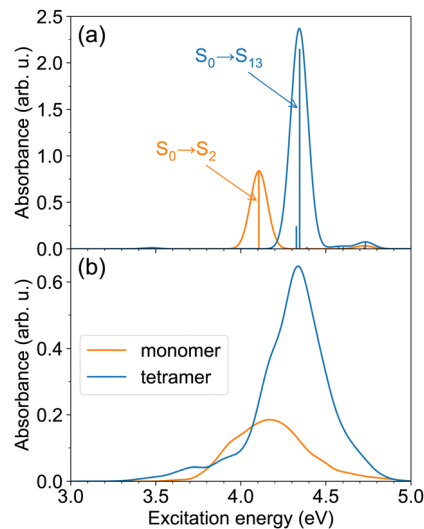


Fig. 2 (a) The $\pi\pi^*$ absorption bands of the monomer and the tetramer, calculated with TD- ω B97X-D/def2-SV(P), for single model geometries. The monomer geometry was optimized at the B3LYP/def2-SV(P) level; the tetramer geometry (see Fig. 1a) was constructed from the optimized monomer geometry. (b) The $\pi\pi^*$ absorption bands of the monomer and the tetramer at $T \approx 300$ K. The spectra are obtained using TD- ω B97X-D/def2-SV(P) calculations for 100 snapshots.

~ 0.01 . This transition is red-shifted by ~ 0.64 eV with respect to the peak of the monomeric $\pi\pi^*$ band.

The spectra at ~ 300 K are shown in Fig. 2b. The spectra are broader in comparison to those for the model geometries in Fig. 2a due to nuclear motion. The tetramer absorption band with a peak at 4.34 eV is blue-shifted by ~ 0.17 eV with respect to the monomer absorption maximum located at 4.17 eV. The maximum tetramer absorbance is ~ 3.5 times larger than the maximum monomer absorbance. Apart from the intense blue-shifted band, the tetramer spectrum exhibits a tail extending in the red direction, up to ~ 3.3 eV. This demonstrates that the spectrum of the H-aggregate is more complex than merely a blue-shifted absorption band, and that the aggregate may be excited not only at energies above the monomer absorption, but also below it.

We turn now to the question of exciton delocalization. Fig. 3a shows the FTDM matrix for the $S_0 \rightarrow S_{13}$ transition of the tetramer, for the model geometry of Fig. 1a. The exciton is mostly composed of local excitations, whereas CT excitations represent a minor contribution. Specifically, the sum of diagonal elements of the FTDM matrix is 86%, whereas the sum of off-diagonal elements is 14%, as shown in Fig. 3b. Furthermore, we also show the diagonal F_{XX} values in Fig. 3b. They demonstrate that the exciton is delocalized over the tetramer, with two molecules in the middle bearing $\sim 24\%$ of excitation each, and two end molecules $\sim 19\%$ each.

Furthermore, we calculated the FTDM matrix for the brightest transition of each of 100 MD snapshots (see distributions of the brightest state labels and corresponding oscillator strengths in Fig. S6 and S7 (ESI[†]), respectively). The averaged \mathbf{F} matrix at ~ 300 K is shown in Fig. 3c. We again see that the local



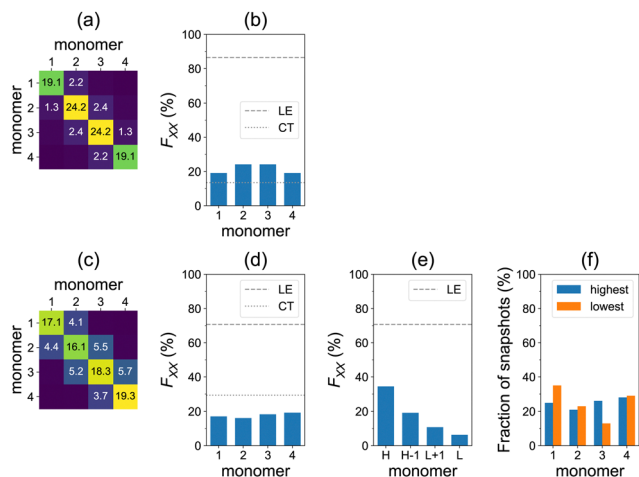


Fig. 3 Top row: Calculations for the single model geometry, for the brightest transition ($S_0 \rightarrow S_{13}$). (a) The \mathbf{F} matrix. (b) Diagonal F_{xx} elements and the sum of diagonal (LE) and off-diagonal (CT) elements. Bottom row: Calculations for the ensemble of 100 snapshots, for the brightest $\pi\pi^*$ transitions. (c) The \mathbf{F} matrix averaged over all snapshots. (d) Diagonal F_{xx} elements of the averaged \mathbf{F} matrix and the sum of diagonal (LE) and off-diagonal (CT) elements. (e) The averaged highest to lowest diagonal values and their sum (LE). (f) Relative frequency with which monomers have the highest or lowest F_{xx} value.

excitations are dominant, but CT excitations are more pronounced than for the single model geometry (compare with Fig. 3a). Namely, we find 71% LE and 29% CT in total (Fig. 3d). The diagonal elements of the averaged \mathbf{F} matrix range from ~ 16 to $\sim 19\%$, showing exciton delocalization for the ensemble (Fig. 3d).

However, this delocalization represents an averaged picture only. Looking at Fig. 3c and d, it is not possible to conclude about the extent of delocalization for individual members of the ensemble. Therefore, we perform the additional analysis that identifies the highest to lowest monomers for each individual snapshot (as described above). The averaged highest to lowest FTDM values are shown in Fig. 3e. The H fragment acquires 34% of excitation, whereas the L fragment only 6%. The $H-1$ and $L+1$ carry 19% and 11%, respectively. Thus, the conformational disorder leads to a partial localization of the bright excitons in the studied azobenzene aggregate. Each of the four monomers may be the highest (or lowest) at that (Fig. 3f), depending on a given MD snapshot.

We have also calculated the spectra using the ω B97 functional³⁵ (with no short-range exact exchange in contrast to ω B97X-D) to check the sensitivity of the results to a given functional. Qualitatively, the findings are the same as with ω B97X-D. Quantitatively, at the TD- ω B97 level, the spectra are blue-shifted (Fig. S4 and S5, ESI[†]), charge-transfer elements of FTDM are smaller (Fig. S8, ESI[†]), and the highest fragment acquires 53% of excitation (instead of the above-mentioned 34% on the TD- ω B97X-D level).

Next, we have also considered dynamics corresponding to much lower temperatures. To do so, we launched a constant-energy trajectory with zero initial velocities. The resultant temperatures are less than 26 K (Fig. S3, ESI[†]). Expectedly,

spectral broadening is smaller than that at 300 K (Fig. S4, ESI[†]). Importantly, the brightest excitons are more delocalized at low temperatures, as revealed by the highest/lowest FTDM analysis (Fig. S9 and S10, ESI[†]).

The $\pi\pi^*$ absorption band of the tetramer at ~ 300 K possesses a tail in the low-energy range, at 3.3–4.0 eV (Fig. 2b), as mentioned above. We analyzed the exciton localization for the $S_0 \rightarrow S_5$ transition, which is the lowest energy $\pi\pi^*$ transition and is located at the $\pi\pi^*$ absorption onset (see SI2.2, ESI[†]). At the model tetramer geometry, this transition is predominantly delocalized over the central dimer (Fig. S12, ESI[†]). It also shows sizable CT excitations between the middle molecules. MD at ~ 300 K again leads to considerable single-geometry localization, whereas the excitons are more delocalized at low temperatures (< 26 K in our case), see SI2.2 (ESI[†]). In this respect, it is interesting to note that photoisomerization in the SAMs (if it occurs) is believed to be triggered by localized excitations.³⁶

Furthermore, we turn to the lower energy $n\pi^*$ band. The $S_0 \rightarrow S_1$ transition of the *trans* monomer is a dark transition (*i.e.*, a transition with zero oscillator strength) located at ~ 2.55 eV. The four lowest transitions in the tetramer ($S_0 \rightarrow S_1$ to $S_0 \rightarrow S_4$), which originate from the lowest monomeric transition, are also dark and reside at ~ 2.43 – 2.52 eV. [The given excitation energies are obtained with TD- ω B97X-D/def2-SV(P) at the B3LYP/def2-SV(P) geometries.] The $n\pi^*$ transitions become allowed (although they carry a weak oscillator strength) due to structural distortions,³⁷ realized in the MD simulations. The $n\pi^*$ absorption bands of the monomer and the tetramer at ~ 300 K are shown in Fig. 4. Both exhibit a maximum at 2.53–2.54 eV. The tetramer absorbance at the maximum is ~ 2.3 times larger than the monomer absorbance.

Furthermore, the lowest four transitions of the aggregate, at the model geometry of Fig. 1a, are dominated by local excitations as can be seen in Fig. 5a–d. The $S_0 \rightarrow S_1$ and $S_0 \rightarrow S_2$ transitions are predominantly delocalized over the dimer in the middle, whereas the $S_0 \rightarrow S_3$ and $S_0 \rightarrow S_4$ transitions are mostly delocalized over the terminal molecules.

The averaged \mathbf{F} matrix for the brightest transition within the $n\pi^*$ band is shown in Fig. 5e. Again, we see that the exciton is delocalized in the averaged picture (Fig. 5f). However, the analysis based on sorting diagonal values for individual

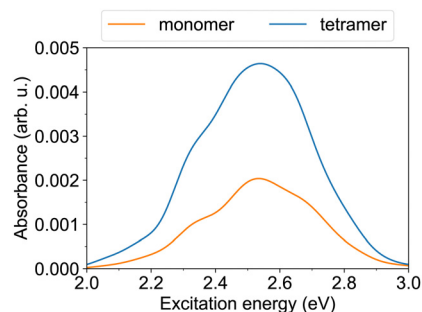


Fig. 4 The $n\pi^*$ absorption bands of the monomer and the tetramer at $T \approx 300$ K. The spectra are obtained using TD- ω B97X-D/def2-SV(P) calculations for 100 snapshots.



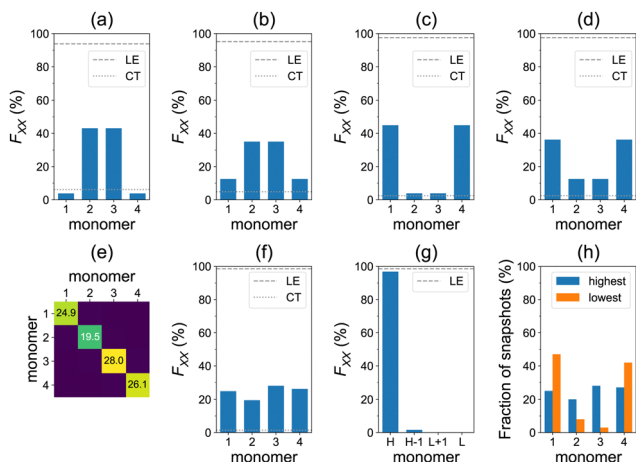


Fig. 5 Top row: Calculations for the single model geometry, for the lowest four transitions ($S_0 \rightarrow S_1$ to $S_0 \rightarrow S_4$). Diagonal F_{XX} elements and the sum of diagonal (LE) and off-diagonal (CT) elements are shown for (a) $S_0 \rightarrow S_1$, (b) $S_0 \rightarrow S_2$, (c) $S_0 \rightarrow S_3$, and (d) $S_0 \rightarrow S_4$ transitions. Bottom row: Calculations for the ensemble of 100 snapshots, for the brightest $\pi\pi^*$ transitions. (e) The \mathbf{F} matrix averaged over all snapshots. (f) Diagonal F_{XX} elements of the averaged \mathbf{F} matrix and the sum of diagonal (LE) and off-diagonal (CT) elements. (g) The averaged highest to lowest diagonal values and their sum (LE). (h) Relative frequency with which monomers have the highest or lowest F_{XX} value.

snapshots reveals a pronounced exciton localization ($F_H = 97\%$) to a single azobenzene unit (Fig. 5g). This strong localization can be explained by a weak exciton coupling of the $\pi\pi^*$ states.³⁸ Each monomer may take a role of the highest monomer (Fig. 5h), which results in the delocalized averaged pictures of Fig. 5e and f. We also note that the CT contributions are negligible in this case (Fig. 5f). At low temperatures, the $\pi\pi^*$ excitons are less localized ($F_H = 74\%$), see Fig. S18 (ESI[†]). The ω B97 results are very similar: $F_H = 97\%$ at ~ 300 K (Fig. S17, ESI[†]) and $F_H = 75\%$ at < 26 K (Fig. S19, ESI[†]).

In addition, we performed dynamics calculations employing ω B97X-D instead of B3LYP+D3 to check the effect of the functional used in step (i). The results are presented in S13

(ESI[†]). While qualitatively we can draw the same conclusions, there are some quantitative differences, most noticeably for the spectra and the ensemble-averaged delocalization of the $\pi\pi^*$ band.

Furthermore, to account for both LE and CT contributions in analysis of single-geometry (de)localization, we computed inverse participation ratio (IPR) (also known as participation number (PN) or delocalization length (DL)), which is defined (for a certain excited state at a given geometry) as:³⁹

$$\text{IPR} = \frac{1}{\sum_X \left(\frac{\sum_Y F_{XY} + F_{YX}}{2} \right)^2} \quad (2)$$

IPR quantifies (de)localization by a scalar (in general noninteger) value ranging from 1 (complete localization) to 4 (complete delocalization). The mean value of IPR for an ensemble of N_s

snapshots is then $\frac{\sum \text{IPR}_\alpha}{N_s}$. IPR for the model geometry and mean IPR with standard deviation for ensembles of 100 snapshots for the studied excitons are shown in Table 1. The IPR values for individual snapshots are provided in Fig. S11, S16, S20, S29, S34 and S39 (ESI[†]). As can be seen from Table 1, IPRs corroborate the findings of the highest/lowest FTDM analysis discussed above.

Finally, to shed light on the strength of exciton–vibration coupling, we calculated time evolution of IPR and F_H along with geometrical changes for the first 100 fs of the high temperature (~ 300 K) MD trajectory (Fig. S40, ESI[†]). Fast uncorrelated intramolecular vibrations lead to rapid localization of excitonically weakly coupled $\pi\pi^*$ states, whereas excitonically strongly coupled $\pi\pi^*$ states undergo only moderate changes in (de)localization.

In summary, we studied the effect of conformational disorder on the exciton states of a model azobenzene tetramer by means of first-principles calculations. The ground-state MD was performed using dispersion-corrected DFT functionals, and the excited states were computed with long-range corrected functionals. Using transition density matrix analysis, we found that the conformational disorder at ~ 300 K induces partial

Table 1 Inverse participation ratios (IPRs) for the studied excitons^a

Model geometry ^b			B3LYP+D3 dynamics		ω B97X-D dynamics	
TD- ω B97X-D	TD- ω B97	T (K)	TD- ω B97X-D	TD- ω B97	TD- ω B97X-D	TD- ω B97
Brightest $\pi\pi^*$ transition						
3.91 [$S_0 \rightarrow S_{13}$]	3.97 [$S_0 \rightarrow S_{10}$]	~ 300 < 26	3.1 (0.5) 3.8 (0.1)	2.5 (0.6) 3.8 (0.2)	2.9 (0.6) 3.8 (0.2)	2.5 (0.6) 3.8 (0.2)
$S_0 \rightarrow S_5$ ($\pi\pi^*$) transition						
2.94 [$S_0 \rightarrow S_5$]	2.88 [$S_0 \rightarrow S_5$]	~ 300 < 26	2.1 (0.5) 2.7 (0.2)	1.9 (0.5) 2.6 (0.2)	2.1 (0.4) 2.7 (0.2)	1.9 (0.4) 2.7 (0.2)
Brightest $n\pi^*$ transition						
2.40 [$S_0 \rightarrow S_1$]	2.41 [$S_0 \rightarrow S_1$]	~ 300	1.1 (0.1)	1.1 (0.2)	1.1 (0.2)	1.1 (0.2)
3.29 [$S_0 \rightarrow S_2$]	3.42 [$S_0 \rightarrow S_2$]	< 26	1.8 (0.8)	1.7 (0.8)	1.9 (0.8)	1.8 (0.8)
2.39 [$S_0 \rightarrow S_3$]	2.40 [$S_0 \rightarrow S_3$]	—	—	—	—	—
3.28 [$S_0 \rightarrow S_4$]	3.41 [$S_0 \rightarrow S_4$]	—	—	—	—	—

^a Corresponding transitions for the model geometry are shown in square brackets. IPR standard deviations are shown in parentheses. ^b Tetramer geometry is constructed from the B3LYP/def2-SV(P) optimized monomer geometry.



localization of the brightest, $\pi\pi^*$ excitons, whereas the excitons at $T < 26$ K are more delocalized. The $n\pi^*$ excitons undergo a strong localization (to a single monomer) at ~ 300 K. At low temperatures (< 26 K) the effect is weakened, but the $n\pi^*$ localization is still pronounced.

Future research will be devoted to the investigation of the exciton dynamics in azobenzene aggregates by means of non-adiabatic mixed quantum-classical MD simulations.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The author thanks Peter Saalfrank for providing computational resources, for his comments on the manuscript and for his support. The author thanks Foudhil Bouakline for valuable discussions. The author also thanks the anonymous reviewer for useful comments on the manuscript. The author thanks the Deutsche Forschungsgemeinschaft (DFG, German Research Foundation) for financial support—project number 454020933. (Gefördert durch die Deutsche Forschungsgemeinschaft (DFG)—Projekt Nummer 454020933.)

Notes and references

‡ In general, for a nonlinear polyatomic molecule with N atoms, temperature is related to the kinetic energy as $E_{\text{kin}} = (3N - 6)\frac{1}{2}kT$.²¹ However, in our models the four lowest hydrogen atoms are fixed and thus $E_{\text{kin}} = (3N - 12 - 1)\frac{1}{2}k\tilde{T}$ (here, 12 corresponds to frozen coordinates of four fixed atoms and 1 accounts for the rigid-body rotation around the axis going through the fixed atoms). Therefore, $\tilde{T} = \frac{3N - 6}{3N - 13}T$, which results in $\tilde{T} \approx 307.6$ K for $T = 300$ K ($N = 96$).

- 1 A. S. Davydov, *Sov. Phys. Usp.*, 1964, **7**, 145–178.
- 2 M. Kasha, H. R. Rawls and M. A. El-Bayoumi, *Pure Appl. Chem.*, 1965, **11**, 371–392.
- 3 N. J. Hestand and F. C. Spano, *Chem. Rev.*, 2018, **118**, 7069–7163.
- 4 T. Mirkovic, E. E. Ostroumov, J. M. Anna, R. van Grondelle, Govindjee and G. D. Scholes, *Chem. Rev.*, 2017, **117**, 249–293.
- 5 G. J. Hedley, A. Ruseckas and I. D. W. Samuel, *Chem. Rev.*, 2017, **117**, 796–837.
- 6 A. Goulet-Hanssens, F. Eisenreich and S. Hecht, *Adv. Mater.*, 2020, **32**, 1905966.
- 7 C. Gahl, R. Schmidt, D. Brete, E. R. McNellis, W. Freyer, R. Carley, K. Reuter and M. Weinelt, *J. Am. Chem. Soc.*, 2010, **132**, 1831–1838.
- 8 E. Titov, A. Sharma, N. Lomadze, P. Saalfrank, S. Santer and M. Bekir, *ChemPhotoChem*, 2021, **5**, 926–932.
- 9 D. T. Valley, M. Onstott, S. Malyk and A. V. Benderskii, *Langmuir*, 2013, **29**, 11623–11631.
- 10 E. Titov, G. Granucci, J. P. Götze, M. Persico and P. Saalfrank, *J. Phys. Chem. Lett.*, 2016, **7**, 3591–3596.

- 11 V. Cantatore, G. Granucci, G. Rousseau, G. Padula and M. Persico, *J. Phys. Chem. Lett.*, 2016, **7**, 4027–4031.
- 12 T. E. Dykstra, E. Hennebicq, D. Beljonne, J. Gierschner, G. Claudio, E. R. Bittner, J. Knoester and G. D. Scholes, *J. Phys. Chem. B*, 2009, **113**, 656–667.
- 13 W. Barford and D. Trembath, *Phys. Rev. B*, 2009, **80**, 165418.
- 14 T. Nelson, S. Fernandez-Alberti, A. E. Roitberg and S. Tretiak, *Phys. Chem. Chem. Phys.*, 2013, **15**, 9245–9256.
- 15 L. Alfonso Hernandez, T. Nelson, M. F. Gelin, J. M. Lupton, S. Tretiak and S. Fernandez-Alberti, *J. Phys. Chem. Lett.*, 2016, **7**, 4936–4944.
- 16 D. Ondarse-Alvarez, T. Nelson, J. M. Lupton, S. Tretiak and S. Fernandez-Alberti, *J. Phys. Chem. Lett.*, 2018, **9**, 7123–7129.
- 17 E. Titov, A. Humeniuk and R. Mitrić, *Phys. Chem. Chem. Phys.*, 2018, **20**, 25995–26007.
- 18 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 5648–5652.
- 19 P. J. Stephens, F. J. Devlin, C. F. Chabalowski and M. J. Frisch, *J. Phys. Chem.*, 1994, **98**, 11623–11627.
- 20 S. Grimme, S. Ehrlich and L. Goerigk, *J. Comput. Chem.*, 2011, **32**, 1456–1465.
- 21 P. H. Hünenberger, in *Thermostat Algorithms for Molecular Dynamics Simulations*, ed. C. Holm and K. Kremer, Springer Berlin Heidelberg, Berlin, Heidelberg, 2005, pp. 105–149.
- 22 J.-D. Chai and M. Head-Gordon, *Phys. Chem. Chem. Phys.*, 2008, **10**, 6615–6620.
- 23 A. Dreuw and M. Head-Gordon, *J. Am. Chem. Soc.*, 2004, **126**, 4007–4016.
- 24 E. Titov, *Molecules*, 2021, **26**, 4245.
- 25 E. Titov and P. Saalfrank, *J. Phys. Chem. A*, 2016, **120**, 3055–3070.
- 26 F. Weigend and R. Ahlrichs, *Phys. Chem. Chem. Phys.*, 2005, **7**, 3297–3305.
- 27 M. J. Frisch *et al.*, *Gaussian 16 Revision C.01*, Gaussian Inc., Wallingford CT, 2016.
- 28 S. Tretiak and S. Mukamel, *Chem. Rev.*, 2002, **102**, 3171–3212.
- 29 F. Plasser and H. Lischka, *J. Chem. Theory Comput.*, 2012, **8**, 2777–2789.
- 30 A. A. Voityuk, *J. Chem. Phys.*, 2014, **140**, 244117.
- 31 S. Mai, F. Plasser, J. Dorn, M. Fumanal, C. Daniel and L. González, *Coord. Chem. Rev.*, 2018, **361**, 74–97.
- 32 T. Lu and F. Chen, *J. Comput. Chem.*, 2012, **33**, 580–592.
- 33 D. Ondarse-Alvarez, S. Kömürlü, A. E. Roitberg, G. Pierdominici-Sottile, S. Tretiak, S. Fernandez-Alberti and V. D. Kleiman, *Phys. Chem. Chem. Phys.*, 2016, **18**, 25080–25089.
- 34 C. Rietze, E. Titov, G. Granucci and P. Saalfrank, *J. Phys. Chem. C*, 2020, **124**, 26287–26295.
- 35 J.-D. Chai and M. Head-Gordon, *J. Chem. Phys.*, 2008, **128**, 084106.
- 36 W. Bronsch, T. Moldt, L. Boie, C. Gahl and M. Weinelt, *J. Phys.: Condens. Matter*, 2017, **29**, 484002.
- 37 T. Cusati, G. Granucci, M. Persico and G. Spighi, *J. Chem. Phys.*, 2008, **128**, 194312.
- 38 E. Titov, T. Kopp, J. Hoche, A. Humeniuk and R. Mitrić, *Phys. Chem. Chem. Phys.*, 2022, **24**, 12136–12148.
- 39 J. J. Nogueira, F. Plasser and L. González, *Chem. Sci.*, 2017, **8**, 5682–5691.

