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Tailoring the molecular weight of polymer additives for organic semiconductors

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A binary system comprising both an organic semiconductor and a polymer additive has attracted extensive research interests due to its great potential for use in high-performance, solution-processable electronic devices on flexible substrates. The molecular weight of polymer additives plays a critical role in modulating the crystal growth, enabling phase segregation, optimizing thin film morphology, and improving the charge transport of organic semiconductors. Here, we provide an in-depth review of the recent progress in studying amorphous and semicrystalline polymeric additives, including polystyrene, poly(α -methylstyrene), polymethyl methacrylate, and polyethylene oxide, and fully discuss the effect of the different polymer molecular weights on semiconductor crystallization, active layer composition, and the electrical performance of miscellaneous organic semiconductors. Using the representative examples of 6,13-bis(triisopropylsilyl)ethynylpentacene (TIPS pentacene), 2,7-dioctyl[1]benzothieno[3,2-*b*][1]benzothiophene (C₈-BTBT), and 2,8-difluoro-5,11-bis(triethylsilyl)ethynylanthradithiophene (diF-TES-ADT), this work sheds light on utilizing these universal polymers with varying molecular weights to powerfully manipulate the charge transport of other high-mobility, solution-processable organic semiconductors.

1. Introduction and background

1.1 Recent progress in flexible electronics

In recent years, research on flexible electronics has witnessed rapid progress.^{1–11} Great efforts have been made to study solvent

choices, controllable crystallization, charge carrier mobilities, and organic electronic device applications of solution-processed small-molecular organic semiconductors, such as 6,13-bis(triisopropylsilyl)ethynylpentacene (TIPS pentacene),^{12–17} 2,7-dioctyl[1]benzothieno[3,2-*b*][1]benzothiophene (C₈-BTBT),^{18–21} 2,5-di-(2-ethylhexyl)-3,6-bis(5'-*n*-hexyl-2,2',5',5''-terthiophen-5-yl)-pyrrolo-[3,4-*c*]pyrrole-1,4-dione (SMDPPEH)^{22–24} and 2,8-difluoro-5,11-bis(triethylsilyl)ethynylanthradithiophene (diF-TES-ADT).^{25–28} For instance, based on the structural engineering of the organic semiconductor pentacene, the herringbone packing patterns in TIPS pentacene are interrupted by the attachment of the bulky side chains, which improves the solubility of TIPS pentacene in

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organic solvents.^{29,30} The enhanced π - π stacking in TIPS pentacene is attributed to higher charge transport.³¹ These research endeavors have opened up vast opportunities for organic semiconductors to be applied in high performance organic electronic devices such as organic thin film transistors,^{32–35} organic gas sensors,^{36–38} photovoltaic devices^{39–41} and logic circuits.^{42–44} In this section, we will review the important studies that investigated the solution processability and charge carrier mobility of organic semiconductor based thin film transistors, as well as their application in fabricating organic gas sensors.

First, research studies have been devoted to exploring the solvent possibilities of organic semiconductors, allowing the modulation of semiconductor morphology and charge transport.^{45–47} For instance, Kim *et al.* investigated the effect of different solvents on the thin film morphology and crystallinity of TIPS pentacene.⁴⁸ Solvents with high boiling points such as chlorobenzene and xylene resulted in dendritic morphology with higher crystallinity, whereas solvents with low boiling points such as chloroform led to an amorphous film with low crystallinity. Choi *et al.* studied the correlation between the solvent boiling point, grain size and charge transport.²⁹ Spin coating TIPS pentacene from chlorobenzene with a high boiling point produced crystals with a large grain size and high crystallinity, yielding a mobility that is 5 orders of magnitude higher than that from a low boiling point solvent such as chloroform. Hwang *et al.* reported the effect of different solvents including chlorobenzene and tetralin on the vertical phase segregation and compositional structure of TIPS pentacene/polymer blends.⁴⁹ Distinct phase segregation and enhanced crystallization were observed based on the solvent tetralin, attributed to higher mobility values. Ozorio *et al.* discovered how the different solvent choices impacted the vertical phase segregation and charge transport in the TIPS pentacene/poly(3-hexylthiophene) (P3HT) blends.⁵⁰ Solvent trichlorobenzene caused a moderate vertical phase segregation between TIPS pentacene and P3HT and gave rise to optimized TIPS pentacene film morphology and enhanced P3HT ordering, which yielded an output current that is 39 times larger as compared to the

counterpart based on other solvents such as chloroform and toluene.

Besides, highly effective alignment methods have been shown to enhance the charge carrier mobilities of the organic semiconductors close to or over $10 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. While miscellaneous alignment methods have been reported, they can be mainly summarized into the categories of blading coating, substrate patterning, slot-die coating and controlled evaporation based methods. First, blade coating based methods involve either a running blade over a stationary substrate or a stationary blade over a moving substrate.^{51–54} The resultant film thickness is determined by the gap between the blade and substrate, the coating speed and the viscoelastic properties of the solution. Solution shearing, as an enhanced version of the blade coating method, involves the use of a tilted blade to deposit the organic semiconductor solutions.^{55–59} Solution shearing has been reported to effectively tune the molecular packing, manipulate the crystal alignment and control the thin film morphology of organic semiconductors.^{60,61} Peng *et al.* demonstrated a “dual solution-shearing” method and reported a mobility of $10 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ using the organic crystals of 2,7-didecyl[1]benzothieno[3,2-*b*][1]benzothiophene (C_{10} -BTBT).⁶² Bilgaiyan *et al.* reported the solution shearing of a 6,6 bis-(*trans*-4-butylcyclohexyl)-dinaphtho[2,1-*b*:2,1-*f'*]thieno[3,2-*b*]thiophene (4H-21DNTT) organic semiconductor based thin film transistor and obtained a mobility of $10.5 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.⁶³ The 4H-21DNTT based CMOS inverter circuit showed sharp switching behaviors with a 31.5 signal gain. Second, substrate patterning based methods involve the tuning of substrate wettability by using surfactant treatment and/or photolithography patterning, which can effectively confine the deposition of organic semiconductor solutions as well as the subsequent crystal growth and alignment. Zhao *et al.* demonstrated an “asymmetric-wettability topographical template” to control the ordered crystallization of 2,7-dioctyl[1]benzothieno[3,2-*b*][1]benzothiophene (C_8 -BTBT), giving rise to a mobility of $8.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.⁶⁴ Shen *et al.* reported the controlled crystallization and morphologies of the C_8 -BTBT organic semiconductor *via* tuning substrate wettability with soluble polymer films as a modification layer.⁶⁵ The resultant C_8 -BTBT film with optimized grain boundary and domain size exhibited a mobility higher than $7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. Deng *et al.* reported a “channel-restricted meniscus self-assembly (CRMS)” method, which combined photolithography patterning and dip coating, to grow uniform 2,6-diphenylanthracene (DPA) single crystal arrays.⁶⁶ The photoresist patterning confined the lateral growth of DPA single crystals, leading to a mobility of $30.3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. Third, slot die coating based methods deposit the organic semiconductor solution onto a moving substrate directly through a coating head at a tunable rate.^{67–70} Lin *et al.* demonstrated highly ordered single crystalline TIPS pentacene films by using a roll-to-roll slot die coating method, and reported a mobility of $4.2 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for pristine TIPS pentacene.⁷¹ Xie *et al.* reported a roll-to-roll slot die coating method to fabricate thin film transistors.⁷² With Ag nanowires deposited by using the R2R coating method as the gate electrode, the C_8 -BTBT/polystyrene



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(PS) based transistor showed a mobility of $18.3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. Fourth, controlled evaporation based methods involve the tuning and control of the solution flow behavior and evaporation rate in order to modulate the deposition and growth of organic semiconductors.^{73–75} Wang *et al.* reported a “Marangoni effect-controlled growth” method to grow a highly oriented C₈-BTBT film with a herringbone stacking structure.⁷⁶ A high mobility of $25 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ was obtained from the C₈-BTBT based transistors incorporating a long channel length. Soeda *et al.* reported the growth of an aligned organic semiconductor 3,11-didecyldinaphtho[2,3-*d*:2',3'-*d'*]benzo[1,2-*b*:4,5-*b'*]dithiophene (C10-DNBDT) with controlled solvent evaporation.⁷⁷ By holding the C10-DNBDT solution at a moving blade edge, continuous solvent evaporation and supply at the same rate gave rise to aligned crystals extending in inches and a mobility of $10 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.

The exceptional high mobilities present great opportunities for the organic semiconductors to be implemented in high performance organic electronic devices such as gas sensors. Manoli *et al.* demonstrated a P3HT thin film transistor based gas sensor for detecting acetone, ethanol and *n*-butanol vapors.⁷⁸ When exposed to the vapors, the P3HT thin film transistors showed a decreased drain current as a result of the enlarged potential barrier at the P3HT grain boundaries as charge trapping. Seo *et al.* reported a 5,11-bis(triethylsilyl-ethynyl)anthradithiophene (TES-ADT) thin film transistor based gas sensor for detecting NO₂ gas.⁷⁹ Exposure to 0 ppm to 100 ppm NO₂ gas caused accumulation of charge carriers and thereby led to an increase in the on-current. Hou *et al.* studied the effect of grain boundaries on the sensing performance of TIPS pentacene transistor based gas sensors.⁸⁰ Solvent vapor annealing with toluene gave rise to a large density of grain boundaries and thereby higher sensitivity when NO₂ gas diffused into the grain boundaries and caused a more pronounced current increase. Shao *et al.* compared the effect of different solvents on the performance of TIPS pentacene transistor based gas sensors.⁸¹ Spin coating TIPS pentacene in solvent *o*-xylene was observed to yield abundant grain boundaries and thereby enhanced gas sensing performance of NO₂.

1.2 Challenges in controlling organic semiconductor crystallization

Despite the major advantages of organic semiconductors such as high mobility and solution processability, the crystallization of organic semiconductors based on solution processing can still be difficult to maneuver.^{82–86} In this section, we employ TIPS pentacene as a benchmark semiconductor to discuss the main challenges that can limit the implementation of organic semiconductors in organic electronic applications. These challenges include crystal misorientation, thermal cracks and grain boundaries, which will be discussed in greater detail below.

First, when drop cast in solution, TIPS pentacene forms dendritic patterns of crystal ribbons, which can be characterized by considerable crystal misorientation and film gaps.^{87–98} In a pivotal study conducted by Chen and coworkers,⁹⁹ the

misorientation angle θ was measured between the TIPS pentacene crystal long axis and the direction from source to drain contact electrodes in the transistor device. This angle was used as the baseline to evaluate the crystal misorientation induced mobility variations. When the smallest angle θ corresponded to the crystal orientation being parallel with the source/drain electrode direction, the largest hole mobility of $0.04 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ was measured from the TIPS pentacene based transistors. When the crystal orientation bridged the source/drain electrode direction at the largest angle θ , the smallest mobility of $0.004 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ was obtained, which showed that the mobility dropped by 10 fold. Similarly, such crystal misorientation has been reported in two other p-type organic semiconductors, *i.e.* 5,6,11,12-tetrachlorotetracene¹⁰⁰ and 2,5-di-(2-ethylhexyl)-3,6-bis(5''-*n*-hexyl-2,2',5',2'') terthiophen-5-yl-pyrrolo[3,4-*c*]pyrrole-1,4-dione (SMDPPEH),¹⁰¹ when grown from a single chloroform solvent *via* the drop casting method.

Besides, the charge transport in the organic semiconductors may be impeded by large-scale thermal cracks, which occur after the semiconductors are exposed to elevated temperature. Chen *et al.* reported that when TIPS pentacene was heated at 150°C , the organic semiconductors started to show thermal cracks,¹⁰² which was linked to the lower surface energy in various planes of the semiconductor. While pristine TIPS pentacene based transistors exhibited a hole mobility of up to $1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ in this study, the counterpart with the occurrence of thermal cracks showed a much lower hole mobility of up to $0.2 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, which counts for a significant reduction in mobility by 80%. The generation of thermal cracks in TIPS pentacene was also reported in a later study by Asaree-Yeboah *et al.* who applied a temperature gradient technique in order to align the crystal growth direction of TIPS pentacene.¹⁰³ These thermal cracks extended to a few hundred micrometers, posing a barrier for efficient charge transport in TIPS pentacene. Despite exerting a blocking effect on charge transport, thermal cracks at the domain boundaries have been reported to generate deep electron traps and enhance the stability of TIPS pentacene based thin film transistor photo responsivity. Cho *et al.* reported that thermal annealing induced cracks at the domain boundary of TIPS pentacene films and related both the stable photoresponsive characteristics and lower charge carrier mobility to the corresponding deep electron traps.¹⁰⁴

Moreover, the crystal growth of organic semiconductors can result in small grain widths.^{105–109} Because crystalline defects and charge trap sites are prevalent at the grain boundaries, small grain widths can be generally associated with a greater amount of defects and traps, which charge carriers must circumvent along the charge transport pathway.^{110–120} If we assume that the channel length L is equal to the length of crystals, or L_G , plus the length of crystal boundary, or L_{GB} , then we have:

$$L = (n - 1)L_{GB} + nL_G \quad (1)$$

where n is the total number of crystals in the channel. If we denote the effective total hole mobility as μ_g and the mobility at the grain boundaries as μ_{GB} , μ_g can be correlated to μ_{GB} based



on the following equation:^{121–123}

$$\frac{L}{\mu_E} = \frac{L - (n-1)L_{GB}}{\mu_G} + \frac{(n-1)L_{GB}}{\mu_{GB}} \quad (2)$$

$$\frac{1}{\mu_E} = \frac{1}{\mu_G} + n \left(\frac{L_{GB}}{L\mu_{GB}} - \frac{L_{GB}}{L\mu_G} \right) \quad (3)$$

The length of crystals can be correlated with the width of grains, or W_G , via the following equation:

$$\sin \theta = \frac{W_G}{L_G} = \frac{nW_G}{L} \quad (4)$$

$$n = \frac{L \sin \theta}{W_G} \quad (5)$$

By assigning $A = \frac{1}{\mu_G}$ and $B = \sin \theta L_{GB} \left(\frac{1}{\mu_{GB}} - \frac{1}{\mu_G} \right)$, we have:

$$\frac{1}{\mu_E} = A + \frac{B}{W_G} \quad (6)$$

Eqn (6) shows a clear correlation between the width of grains and the effective mobility, which states that a small grain width from the solution-processed organic semiconductors can result in lower effective mobility.

2. Benefits of polymer-modulated crystallization

Miscellaneous polymer additives have been extensively studied for mixing with solution processable, small molecular organic semiconductors.^{124–129} When the polymer additives are mixed with semiconducting small molecules, the resultant binary system can benefit from improved semiconductor morphology uniformity,^{130–134} phase segregation-enhanced charge transport^{135–139} and operational device stability.^{140–147} In this section, we will briefly discuss the benefits of mixing organic semiconductors with different categories of polymer additives, including amorphous polymers, conjugated polymers and semicrystalline polymers.

2.1. Amorphous polymers

Mixing organic semiconductors with amorphous polymers can help improve the thin film morphology uniformity and, more importantly, can promote the formation of orientated crystals and reduce crystal misorientations.^{148,149} For example, pristine TIPS pentacene has been reported to form random crystal ribbons or dendritic patterns of crystals by multiple groups.^{150–152} The mixing of TIPS pentacene with amorphous polymer, such as poly(α -methylstyrene) (P α MS), can significantly reduce crystal misorientation.¹⁵³ He *et al.* reported that when drop cast in the solvent of toluene, the TIPS pentacene pristine film exhibited a large misorientation angle of $43.9^\circ \pm 27.8^\circ$, whereas the addition of P α MS with TIPS pentacene strongly prevented such random crystal morphology and reduced the misorientation angle to $2.2^\circ \pm 1^\circ$.³⁰

Another major benefit that rises from blending organic semiconductors with amorphous polymers is the occurrence

of phase segregation between the organic semiconductor and polymer additive.^{154–160} This phase segregation can include lateral phase segregation, vertical phase segregation, or a mixture of both. In particular, vertical phase segregation between the organic semiconductor and amorphous polymer can facilitate the formation of a more concentrated semiconductor layer at the interface between the active layer and the dielectric layer, expediting charge transport of the organic semiconductor. Also, a polymeric encapsulation layer that forms as a result of the vertical phase segregation can protect the semiconductor active layer from being exposed to the ambient environment, enhancing the operational stability of the thin film transistors.^{161,162}

2.2. Conjugated polymers

When organic semiconductors are blended with conjugated polymers, the binary system can give rise to intermolecular interactions between these two components.^{163–168} For example, Chen *et al.* reported that when two different types of conjugated polymers, including P3HT and regiorandom pentacene-bithiophene polymer (PnBT-RRa), were mixed with TIPS pentacene, the resultant binary system exhibited tunable intermolecular interactions that arise from both π - π interactions and hydrophobic interactions. The different extent of molecular structural similarity between TIPS pentacene and the conjugated polymer additives determines the intensity of intermolecular interactions. A small structural similarity between TIPS pentacene and P3HT yielded less intimate mixing and intermolecular interactions, leading to a polymorph with greater changes in the cell parameters. A higher similarity between TIPS pentacene and PnBT-RRa resulted in more intimate mixing, stronger interactions, and a polymorph with smaller changes in the cell parameters. Such differences further modulate the thin film morphology and confine charge transport in the organic semiconductor.⁸⁹ Depending on the different types of conjugated polymers and extent of intermolecular interactions, the TIPS pentacene crystalline morphology can range from straight crystal needles to grass-like ribbons. Simultaneously, the conjugated polymer may exhibit cocrystallization along with the organic semiconductor. Thereby, each conjugated polymer yielded distinct modes of crystallization behavior and charge transport.

2.3. Semicrystalline polymers

When compared to the amorphous and conjugated counterparts, semicrystalline polymers^{169–175} can preserve their own nucleation and crystallization events, which are independent of and also competing with the nuclei formation and crystal growth of the organic semiconductors. In the meanwhile, semicrystalline polymers are still partially amorphous, allowing the modification of diffusivity and surface energies of the organic semiconductor facets similar to amorphous polymers. Unlike conjugated polymers, semicrystalline polymers eliminate the polymer-caused π - π interactions as described above, which may be a complicated process unwanted in some organic semiconductor applications. Thereby, semicrystalline polymers



provide a novel pathway to guide organic semiconductor crystallization and tune its morphology from an unexplored perspective.

3. Overview of polymers with varying molecular weights

In this section, we will provide an overview of the important studies that report mixing organic semiconductors with polymeric additives with different molecular weights, in order to tune the crystal growth, thin film morphology, phase segregation, and charge transport. These polymer additives will be discussed in the following category: P α MS, PS, polymethyl methacrylate (PMMA) and polyethylene oxide (PEO); their molecular structures are shown in Fig. 1(a–d). Fig. 1(e–h) show the molecular structures of the various organic semiconductors mixed with the polymer additives in this section.

3.1 Polystyrene with varying molecular weights

PS is an amorphous and thermoplastic polymer that finds extensive applications in miscellaneous semi-finished products such as plastic films and foams. PS has excellent electrical insulation properties and outstanding optical clarity as a result of its lack of crystallinity. Besides, PS is chemically resistant to diluted bases and acids, but can be attacked by hydrocarbon solvents and UV exposure. The stiffness of the PS polymer backbone results in brittle nature and low impact strength.

Furthermore, PS has a low limit of upper temperature for continuous processing. When mixed with organic semiconductors, PS typically causes pronounced vertical phase segregation,^{129,176} enhances semiconductor wettability,¹⁷⁷ modulates active layer structure and polymorph,¹⁷⁸ reduces charge interfacial traps,¹⁷⁹ and improves charge transport reproducibility.¹⁸⁰

Leonardi *et al.* reported the effect of PS additive with different molecular weights, including 3 K and 100 K, on the vertical phase segregation, thin film morphology and stability of a donor–acceptor random copolymer, *i.e.* PDPP-TT(1)-SVS(9).¹³⁷ AFM images in Fig. 2 show a vertical phase segregation with a bottom PDPP-TT(1)-SVS(9) layer beneath a top PS layer, which could serve as an encapsulation layer and improve the stability of the transistor device. In particular, the PDPP-TT(1)-SVS(9)/PS blend film with the 3 K molecular weight exhibited a non-homogeneous surface, whereas the counterpart with the 100 K molecular weight showed a thicker PS layer with the existence of holes, which resulted from the increased viscosity of the blend solution. PDPP-TT(1)-SVS(9) based transistors with the addition of PS with 3K and 100K molecular weights demonstrated average mobilities of 0.22 cm² V^{−1} s^{−1} and 0.10 cm² V^{−1} s^{−1}, respectively. The enhanced charge transport was a direct benefit from the passivation of surface hydroxyl groups by the vertically-segregated PS polymer additive.

Niazi *et al.* studied the impact of PS additives with different molecular weights on the phase separation, composition and domain size of the organic semiconductor diF-TES-ADT.¹⁸² PS with different molecular weights including 2.2 K and 900 K was mixed with diF-TES-ADT. The mixture was deposited by using the blade coating method with a direction parallel with the source-to-drain electrodes. The thin film morphology of neat diF-TES-ADT and the diF-TES-ADT/PS mixture film is shown in the polarized optical images and AFM images in Fig. 3. It can be inferred that the film morphology and domain size depend on the molecular weight of PS. Besides, both phase segregation and active layer compositions were found to correlate to the molecular weight of PS. In particular, PS with a high molecular weight of 900 K caused the formation of a vertically-segregated bilayer structure: a top diF-TES-ADT layer and a bottom PS layer. Conversely, PS with a low molecular weight of 2.2K led to a mixture of diF-TES-ADT and PS in the top section. A thin film transistor incorporating the diF-TES-ADT/PS mixture as the active layer showed a mobility of up to 6.7 cm² V^{−1} s^{−1} based on the high molecular weight PS, which is 50-times higher than that based on the low molecular weight counterpart. The high performance of the transistor devices was attributed to the single crystalline diF-TES-ADT layer with large domains as a result of vertical phase segregation.

Haase *et al.* reported the addition of PS with different molecular weights in order to reduce the device-to-device mobility variation of the organic semiconductor C₈-BTBT.¹⁸³ Different molecular weights of the PS polymer including 2 K, 20 K, 200 K and 2000 K were tested on the thin film morphology of the organic semiconductor. PS with a low molecular weight resulted in a spherulitic growth of solution-sheared C₈-BTBT films, whereas PS with a larger molecular weight yielded

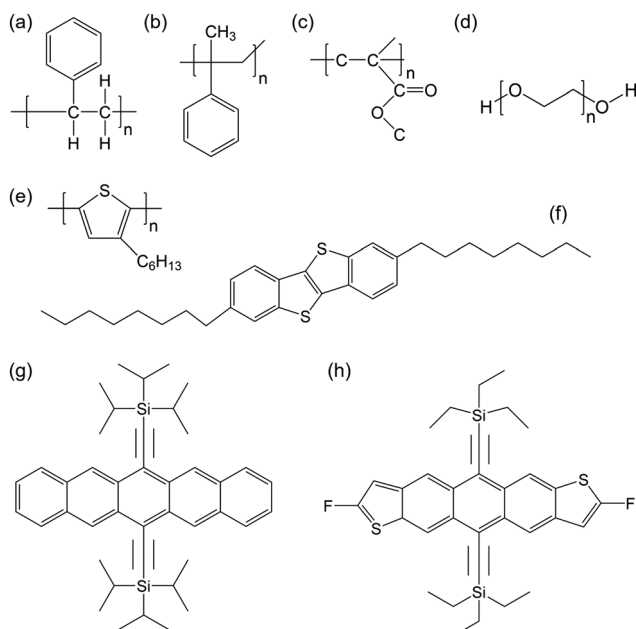


Fig. 1 Molecular structures of the various polymer additives and small-molecular organic semiconductors discussed in this work, including (a) polystyrene (PS), (b) poly(α -methylstyrene) P α MS, (c) polymethyl methacrylate (PMMA), (d) polyethylene oxide (PEO), (e) poly(3-hexylthiophene) (P3HT), (f) 2,7-dioctyl[1]benzothieno[3,2-b][1]benzothiophene (C₈-BTBT), (g) 6,13-bis(triisopropylsilyl)ethynyl pentacene (TIPS pentacene), and (h) 2,8-difluoro-5,11-bis(triethylsilyl)ethynylanthradithiophene (diF-TES-ADT).

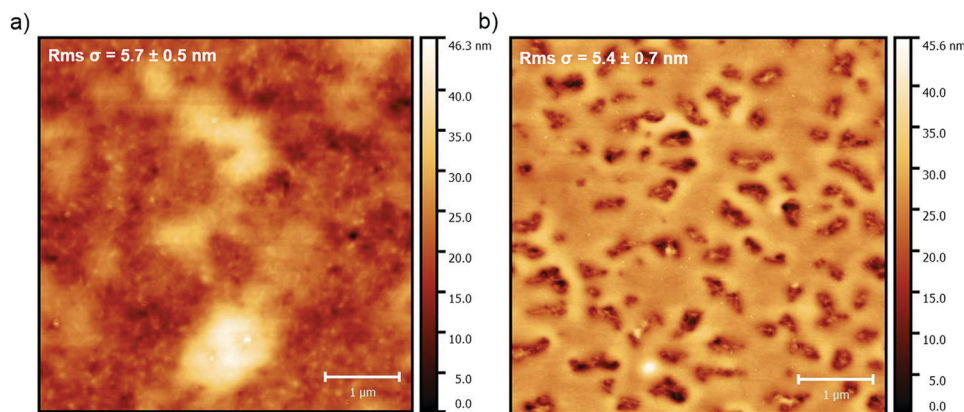


Fig. 2 AFM images showing the topography of the PDPP-TT(1)-SVS(9)/PS mixture film, with different molecular weights of PS including (a) 3 K and (b) 100 K. Reproduced from ref. 181, with permission from Elsevier.

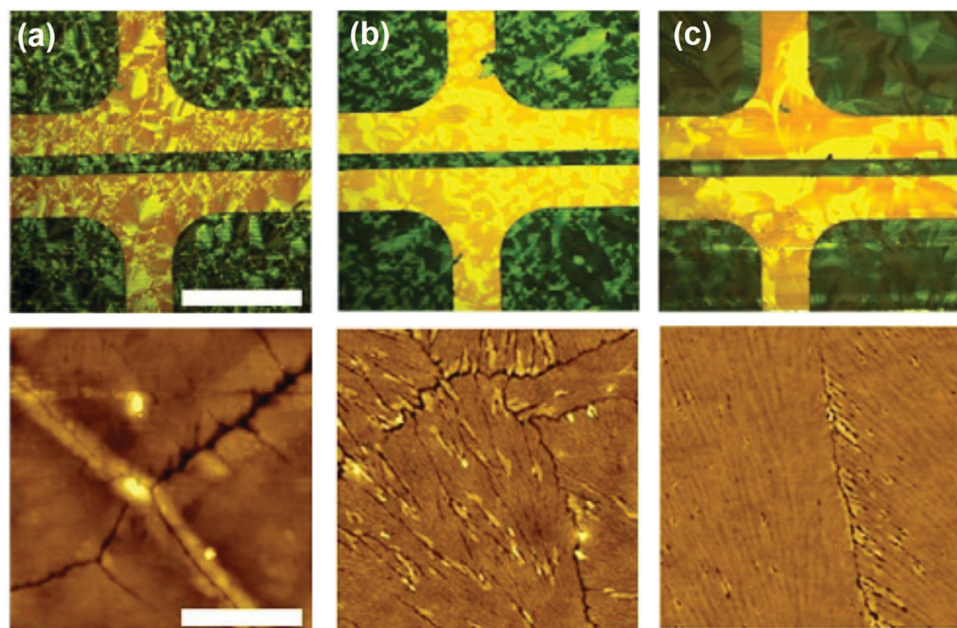


Fig. 3 Polarized optical images and corresponding AFM images of (a) neat diF-TES-ADT, (b) diF-TES-ADT blended with PS with a low molecular weight of 2.2 K, and (c) diF-TES-ADT blended with PS with a high molecular weight of 900 K. The direction of blade coating was parallel with the direction from the source to drain contact electrodes. Reproduced from ref. 182, with permission from Springer Nature.

C₈-BTBT ribbons with enhanced film coverage, as shown in the microscopic optical images in Fig. 4. Accordingly, the highest mobility of 14.4 cm² V⁻¹ s⁻¹ was observed with PS with a molecular weight of 200 K, at an optimized blending ratio of 1 : 1. Finally, distribution of saturation mobility was studied for thin film transistors based on neat C₈-BTBT and the C₈-BTBT/PS mixture in a 1 : 1 ratio. The addition of the PS polymer resulted in a much narrower distribution of mobility as well as reduced standard deviation.

Singsumphan *et al.* studied the effect of PS molecular weight on the polymer viscosity and drop-cast thin film morphology of organic semiconductor 9,10-bis-2-[tris(1-methylethyl)silyl]-ethynyl anthracene (TIPS-anthracene).¹⁸⁴ Fig. 5 shows the

optical images and fluorescence images of drop cast TIPS-anthracene based on mixing with PS of different molecular weights including 13 K, 35 K and 280 K, respectively. For PS with a low molecular weight of 13 K, the mixed TIPS-anthracene exhibited a crystalline morphology of elongated needles. For PS with an intermediate molecular weight of 35 K, the needle shaped TIPS-anthracene crystals adopted alignment in different directions, indicating greater randomness of the thin film morphology. For PS with a high molecular weight of 280 K, the needle-like TIPS-anthracene crystals further suffered from reduced substrate coverage, which was caused by the high viscosity of PS and poor processability. This study indicated that tuning the polymer viscosity by employing polymer



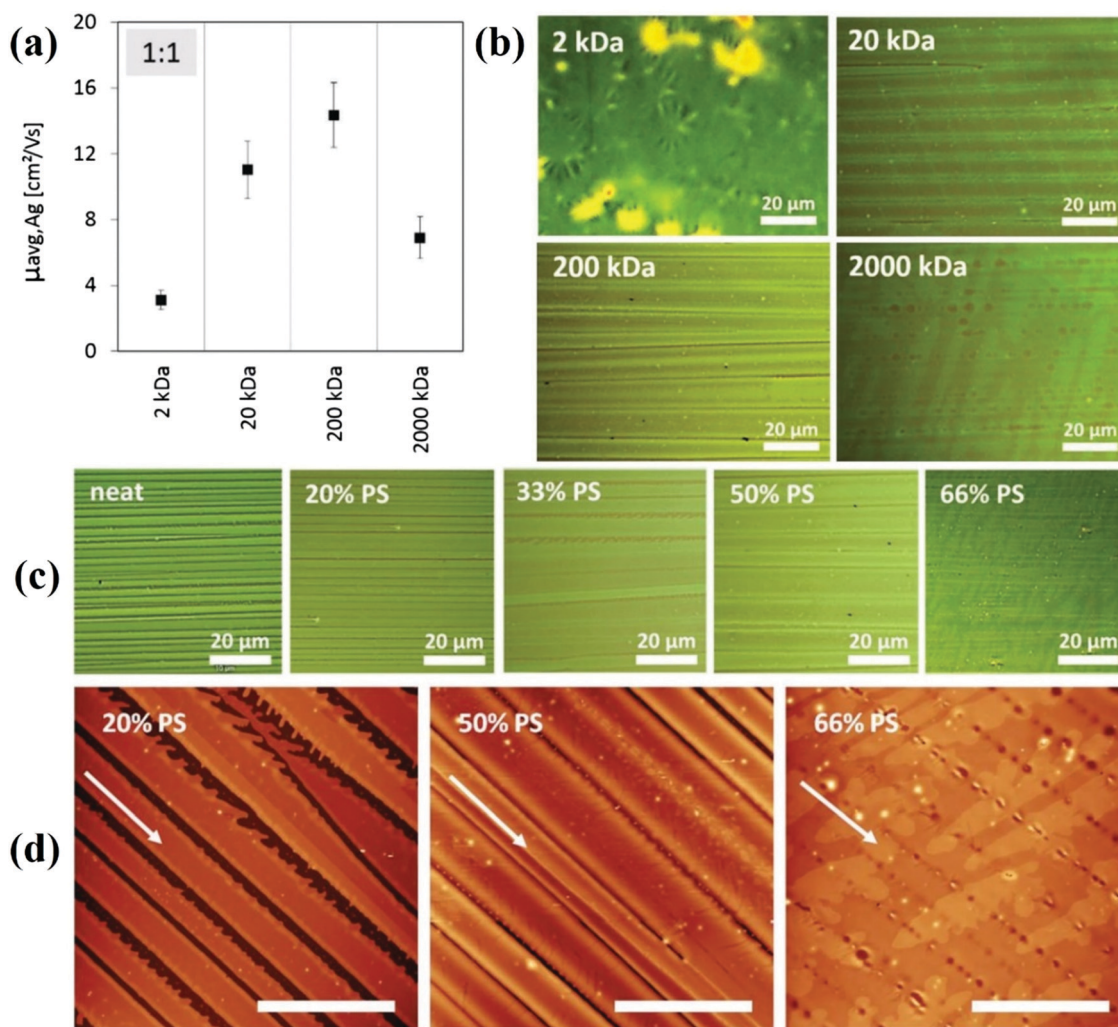


Fig. 4 Microscopic images of solution-sheared C₈-BTBT films from blends based on different PS molecular weights, including 2 K, 20 K, 200 K and 2000 K. Reproduced from ref. 183, with permission from Wiley.

additives with different molecular weights can be a pathway to regulate the thin film morphology of organic semiconductors.

Rocha *et al.* studied the mixing of TIPS pentacene with PS with different molecular weights, including 2 K, 20 K, 200 K and 2000 K, to improve the binary system's miscibility and charge transport.¹⁸⁵ At a mixing ratio of 1:1, the TIPS pentacene/PS mixture was deposited by using a solution shearing method. The resultant crystalline film exhibited different morphologies, dependent on different molecular weights of the PS polymer additive and different solution shearing speed. In particular, highly anisotropic crystalline ribbons and more isotropic spherulitic crystals were obtained based on low and high solution shearing speed, respectively. Electrical characterization results indicated that PS with a higher molecular weight yielded a higher charge carrier mobility as compared to the counterpart with 2 K molecular weight. The low mobility from low molecular weight PS can be attributed to the higher miscibility between PS and TIPS pentacene, preventing the phase segregation and the growth of highly crystalline TIPS pentacene ribbons. On the other hand, PS with a molecular

weight larger than 20 K was observed to induce a phase segregation between TIPS pentacene and PS. The highest mobility of $12.3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ was demonstrated based on the TIPS pentacene/PS mixture with a 20K molecular weight.

Han *et al.* reported the effect of different PS molecular weights on the response of P3HT based gas sensors.¹⁸⁶ Two types of PS, including isotactic semicrystalline PS with molecular weights of 1.3 K, 250 K, 280 K and 290 K and amorphous PS with a molecular weight between 130 K and 290 K, were mixed with P3HT at a 4:1 ratio to form an active layer of the thin film transistors. When the P3HT based transistor devices were exposed to 10 ppm NH₃ gas, the gas sensing response increased as the molecular weight of isotactic semicrystalline PS increased. In contrast, a much lower sensing response was observed for the counterpart based on the amorphous PS polymer additive. The enhanced sensing performance by the isotactic semicrystalline PS with a larger molecular weight was attributed to the formation of vertically-segregated, more continuous P3HT microstructures, which provide an undisrupted pathway for the transport of charge carriers in the transistor



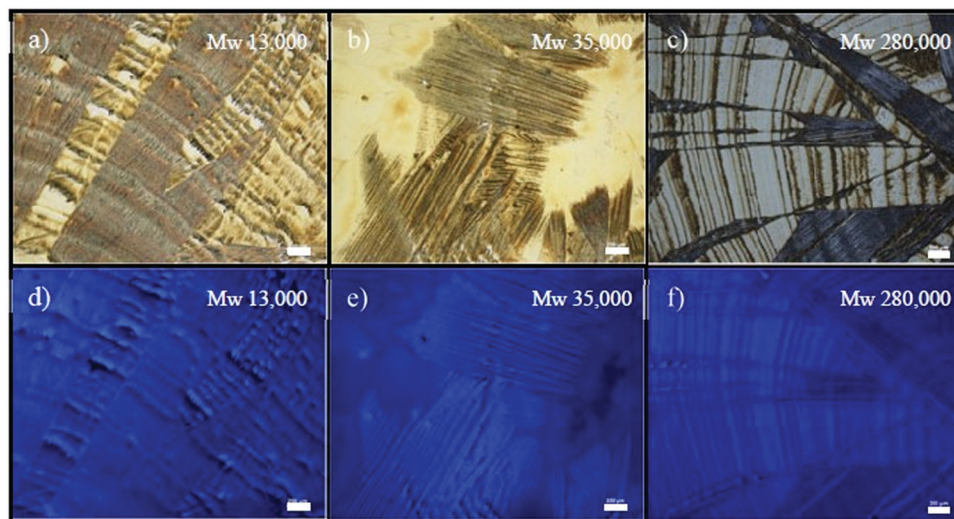


Fig. 5 (a–c) Optical images and (d–f) fluorescence images of TIPS–anthracene/PS blend films based on different PS molecular weights including 13 K, 35 K and 280 K. The scale bar is 200 μm . Reproduced from ref. 184, with permission from Elsevier.

device. A mobility of $0.03 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ was reported from the P3HT/PS based transistors.

3.2 Poly(α -methylstyrene) with varying molecular weights

P α MS is an amorphous polymer with outstanding hardness, heat resistance and solvent resistance properties.¹⁸⁷ While P α MS in general finds good solubility in both aromatic and halogenated solvents, only P α MS with a low molecular weight can be soluble in methyl ethyl ketone. The physical properties of P α MS such as solvent resistance depend on the amount of monomer and low molecular weight polymer in the compositions. Due to its relative brittleness and lower thermal stability (as compared to PS), the molecular weight of P α MS should be kept at a minimum value to enable any molding and extrusion processes to occur. As a result, it is desirable to set the molecular weight of P α MS at a lower limit, in order to render the polymer more useful physical properties. In the field of organic electronics, P α MS has been extensively studied for mixing with various organic semiconductors in order to improve the uniformity of thin film morphology,¹⁸⁸ modulate semiconductor wettability,¹⁸⁹ reduce crystal misorientation,^{30,190} control the formation of thermal cracks,¹⁹¹ enhance charge carrier mobilities,^{192,193} recover electrical performance,¹⁹⁴ and improve the thin film transistor device stability.¹⁹⁵

Ohe *et al.* reported the impact of different molecular weights of the polymeric additive P α MS on phase segregation, thin film morphology and charge carrier mobility of the organic semiconductor TIPS pentacene.¹⁹⁶ The different molecular weights of the polymeric additive P α MS included 2 K, 20 K, 60 K, 100 K and 800 K. Source and drain contact electrodes were prepatterned on the substrate covered with a cross-linked poly(vinyl phenol) gate dielectric layer. Then P α MS was mixed with TIPS pentacene at a 1:1 weight ratio in solution before being spin coated onto the source and drain electrodes. Electrical characterization results showed while the addition of P α MS with a low molecular weight of 2 K resulted in a low mobility, P α MS

with a molecular weight above 20 K yielded a much higher mobility of $0.1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. To study the composition in the vertical profile of the organic semiconductor film, time-of-flight secondary ion mass spectrometry was conducted on the TIPS pentacene/P α MS blends. The secondary Si^- ions and SiH^- ions were used to mark the existence of TIPS pentacene. The addition of a P α MS additive with a molecular weight of 2 K led to a homogeneous distribution of TIPS pentacene, whereas P α MS with a high molecular weight of 100 K promoted a vertical phase segregation between the polymer additive and TIPS pentacene. Furthermore, the polarized optical microscope images showed that the P α MS additive with a high molecular weight of 100 K enlarged the crystalline region on the surface. Flory–Huggins theory, which is indicative of the Gibbs free energy change upon mixing, was applied to understand phase segregation between TIPS pentacene and P α MS. Negative Gibbs free energy of mixing implies homogeneous mixing, whereas positive Gibbs free energy of mixing indicates that phase segregation can more easily occur. The low molecular weight P α MS resulted in a negative value of ΔG_m , indicating it is more likely for the blend film to undergo a homogeneous mixing. In contrast, as the high molecular weight P α MS yielded a positive value of ΔG_m , the vertical phase segregation between the polymer additive and organic semiconductor was promoted.

Kang *et al.* reported the impact of P α MS with different molecular weights on the TIPS pentacene phase segregation behavior and compositional structure by employing the neutron reflectivity method.¹⁹⁷ Two types of P α MS with different molecular weights at 1.3 K and 575 K were blended with TIPS pentacene at a 1:1 weight ratio before being spin coated on the substrate to form an active layer. After this, thermal annealing at 90 $^\circ\text{C}$ was conducted on the active layer. Neutron reflectivity profiles of TIPS pentacene/P α MS blend film are presented in Fig. 6(a and b) based on the different P α MS molecular weights. The addition of P α MS with a lower molecular weight of 1.3 K was observed to induce a vertically segregated film structure,



which is composed of an almost pure TIPS pentacene top layer at the air surface, a P α MS rich layer in the middle section, and a TIPS pentacene rich bottom layer at the substrate interface. The thicknesses from top to bottom layer corresponding to the lower molecular weight P α MS were 177 Å, 326 Å and 70 Å, respectively. In contrast, the mixing of P α MS with a higher molecular weight of 575 K induced a similar vertically segregated film but with different compositions. The thicknesses of the top and bottom TIPS pentacene rich layer based on the higher molecular weight P α MS changed to 134 Å and 117 Å, respectively. To understand the effect of thermal annealing on the crystallinity of the film, grazing-incidence X-ray diffraction (GIXD) was conducted based on P α MS with a molecular weight of 575 K. The GIXD spectra presented in Fig. 6(c and d) showed that thermal annealing resulted in a highly crystalline film with desired crystal orientations, giving rise to a saturation mobility of $0.54 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.

3.3 Poly(methyl methacrylate) with varying molecular weights

PMMA is a thermoplastic polymer with optical transparency. PMMA shows outstanding properties such as high impact strength, scratch resistance, and lightweight. The incorporation of an adjacent methyl group in the PMMA structure inhibits close packing and free rotation, which results in an amorphous nature of PMMA.¹⁹⁸ PMMA has extensive applications such as in optics, polymer viscosity and biomedical engineering, and also finds important implementation in nanotechnology

because of its easy processing and good compatibility with inorganic materials such as carbon nanotubes.¹⁹⁹ As one of the most studied polymers for blending with organic semiconductors, PMMA demonstrates exceptional capability for enhancing semiconductor growth,²⁰⁰ film crystallinity,²⁰¹ grain size,²⁰² phase segregation,²⁰³ device air stability¹⁹⁵ and charge transport.²⁰⁴ Besides, PMMA also shows excellent gate dielectric properties when employed as the polymer gate insulator material for thin film transistors^{205,206} and gas sensors.²⁰⁷

Kim *et al.* studied how different molecular weights of PMMA impacted the polymorphism and charge transport of a C₈-BTBT organic semiconductor.²⁰⁸ PMMA with different molecular weights including 15 K, 120 K, 350 K, and 996 K were mixed with C₈-BTBT before spin coating the blends for crystal growth using a solvent vapor annealing method. As shown in the polarized optical microscopic images in Fig. 7, the crystallization and resultant dimensions of the C₈-BTBT rods, in terms of rod height, width and length, were observed to be dependent on the molecular weight of PMMA. In particular, the rod dimension becomes larger as the PMMA molecular weight increases. This was attributed to the reduced polarity of higher molecular weight PMMA, which allowed the free migration of C₈-BTBT on the PMMA surface to form enlarged dimensions. Electrical characterization results of C₈-BTBT/PMMA (molecular weight 120K) based thin film transistors indicated that the source-drain currents exhibited a tremendous increase by 3–4 orders of magnitude as the device was applied to thermal

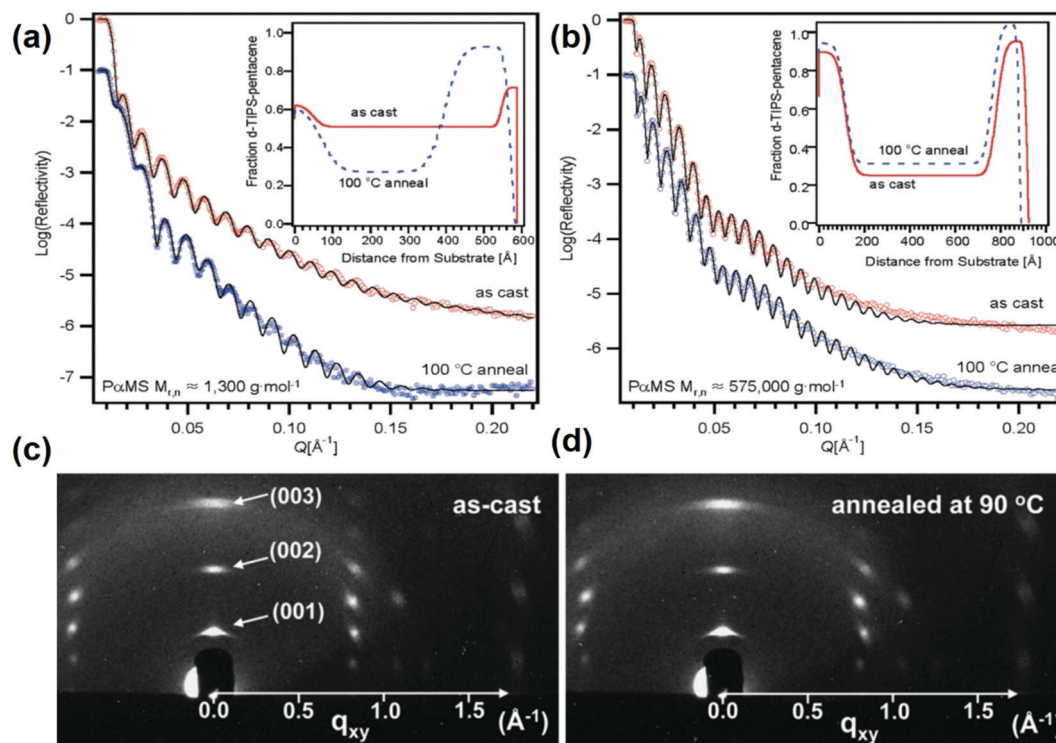


Fig. 6 Neutron reflectivity profiles of the TIPS pentacene/P α MS blend film with different P α MS molecular weights of (a) 1.3 K and (b) 575 K, before and after thermal annealing at 100 °C. The insets show the fitted concentration profiles. GIXD patterns of the TIPS pentacene/P α MS blend film with a molecular weight of 575 K, measured (c) after spin coating and (d) after thermal annealing at 90 °C. Reproduced from ref. 197, with permission from American Chemical Society.



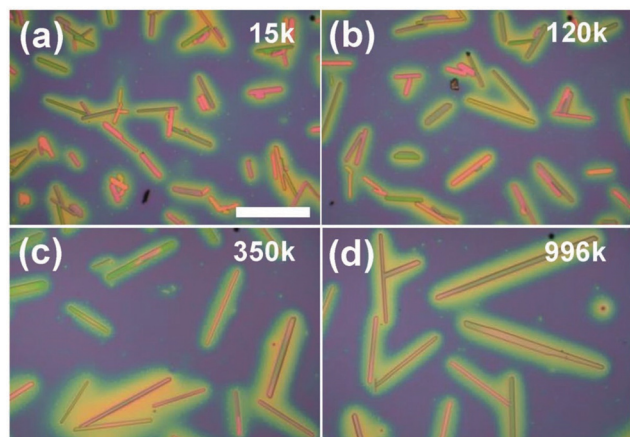


Fig. 7 Polarized optical microscopic images of the crystallization of C_8 -BTBT rods on PMMA with various molecular weights including (a) 15 K, (b) 120 K, (c) 350 K, and (d) 996 K. Reproduced from ref. 208, with permission from Wiley.

cycling. This was due to the increased resistivity of C_8 -BTBT induced by the varying temperatures, at which point the C_8 -BTBT molecules become oriented downward the substrate. A mobility of $15.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ was obtained when heating was applied to the C_8 -BTBT/PMMA blend based thin film transistors.

3.4 Polyethylene oxide with varying molecular weights

Polyethylene oxide (PEO) finds important applications in pharmaceuticals and is compatible with pharmaceutical dosage due to its thermoplastic behavior. When combined with other polymers, PEO can be utilized to realize controllable drug release.²⁰⁹ In particular, PEO with a high molecular weight is suitable for designing melt-extruded matrices with extended release. On the other hand, PEO is a semicrystalline polymer exhibiting unique crystallization behavior in stark contrast to the conjugated and amorphous counterparts. PEO can eliminate the potential impact of conjugated polymer induced π - π interactions and cocrystallization on the target organic semiconductor,³⁸ which is not a feasible process to optimize for simplified device fabrication procedures. With that said, PEO retains the amorphous nature which helps improve the semiconductor diffusivity and surface energies. The semicrystalline PEO is known for its own crystallization, which is independent of but competing with the crystal growth of the organic semiconductor. These traits of PEO uniquely offer a pathway to modulate solution-based organic semiconductor crystallization from a whole new perspective.

He *et al.* studied the effect of semicrystalline polymer additive PEO with different molecular weights including 8 K and 100 K on the crystallization and charge transport of organic semiconductor TIPS pentacene.²¹⁰ With the selection of 8K and 100K molecular weights, the PEO polymer with both light and heavy chain entanglements and its impact on modulating semiconductor crystallization can be compared. PEO was mixed with TIPS pentacene at a 1:1 weight ratio before the mixture was deposited to form the active layer *via* drop casting.

Fig. 8(a–d) show the morphology of the TIPS pentacene/PEO (molecular weight of 8 K) crystals based on different sections of the same substrate. The charge transport direction in the organic crystals is highlighted by red arrows. The dark regions observed in Fig. 8 are the formation of PEO crystals, as a result of its own nucleation and crystallization process. The crystal dimensions of PEO crystals are typically under 20 microns, which is consistent with the observation in Fig. 8. In comparison, TIPS pentacene crystals exhibit clear longitudinal dimension extending to a few hundred of microns and also crystal orientation-induced color variations in the crystalline domains under a polarized light microscope. It can be referred that the TIPS pentacene crystals were aligned by the addition of the PEO polymer with a lower molecular weight of 8 K. The effects on modulating the TIPS pentacene crystallization can be attributed to the partially amorphous nature of PEO, which tunes the semiconductor diffusivity and surface energy of faceted evolution.

In contrast, blending TIPS pentacene with a PEO polymer with a higher molecular weight of 100 K resulted in a distinctly different thin film morphology as shown in the optical images in Fig. 9(a–d). A PEO polymer with a 100 K molecular weight did not benefit TIPS pentacene from reducing random crystal orientations, as indicated by the misoriented red arrows in various directions. Additionally, the film was composed of a top TIPS pentacene/PEO crystal layer as well as a bottom base layer, suggesting strong vertical phase segregation. The blending of PEO with a higher molecular weight likely enhances the interaction between the polymer hydroxyl groups and the silicon dioxide layer on the substrate, further attributing to the phase segregation as observed in the optical images in Fig. 9. Although no obvious crystal alignment was observed, the PEO additive with a molecular weight of 100K showed a more pronounced effect on promoting film formation and increasing

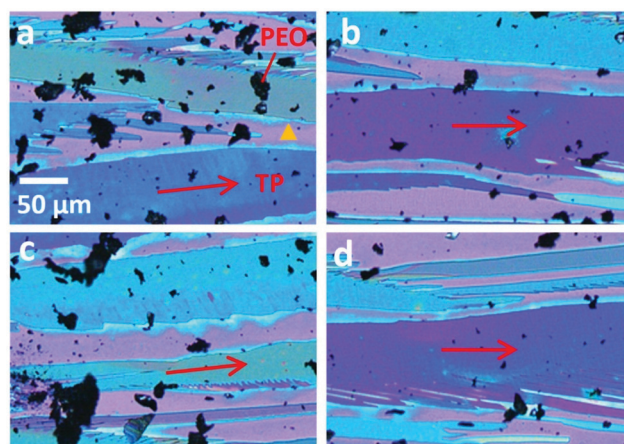


Fig. 8 (a–d) Four polarized microscope images of TIPS pentacene crystals with the PEO polymer additive (molecular weight of 8 K). The charge transport direction in the TIPS pentacene/PEO crystals is highlighted by the red arrows. The substrate uncovered by TIPS pentacene/PEO crystals is marked by the yellow triangles. The scale bar shown in (a) applies to all images. Reproduced from ref. 210, with permission from Springer Nature.



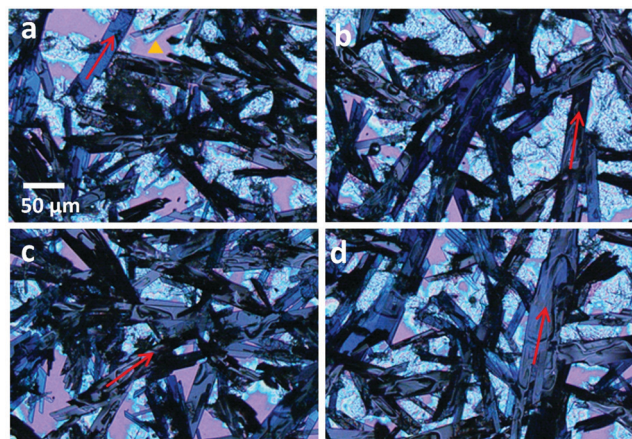


Fig. 9 (a-d) Four polarized optical microscope images of TIPS pentacene crystals with the PEO polymer additive (molecular weight of 100 K). The charge transport direction in the TIPS pentacene/PEO crystals is highlighted by the red arrows. The uncovered substrate is marked by the yellow triangle. The scale bar shown in (a) applies to all images. Reproduced from ref. 210, with permission from Springer Nature.

coverage. TIPS pentacene/PEO mixture film was further used as the active layer for fabricating bottom-gate, top-contact thin film transistors. The addition of PEO with a lower molecular weight of 8 K resulted in a mobility of $0.017 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, which is 5-fold larger as compared to that based on PEO with a molecular weight of 100 K.

The different studies reviewed in this section are summarized in Table 1, which lists the authors, polymer (with its molecular weight), semiconductor, result and mobility.

4. Conclusions and outlook

In this article, we have reviewed the influence of polymer additives with different molecular weights on the crystallization, vertical phase separation, compositional structure, thin film morphology and charge carrier mobility of solution-processed organic semiconductors. We first discussed the general benefits of mixing organic semiconductors with amorphous polymers, conjugated polymers, and semicrystalline polymers, respectively. Then, we reviewed several extensively studied amorphous polymers PS, P α MS and PMMA, as well as semicrystalline polymer PEO, in the context of small-molecular organic semiconductors such as TIPS pentacene, C₈-BTBT and diF-TES-ADT. In particular, PS with different molecular weights impacted solution viscosity, processability, solute intermixing, phase segregation, crystal morphology and active layer compositions. P α MS with different molecular weights influenced Gibbs free energy upon intermixing, phase segregation, semiconductor layer thickness and charge transport. PMMA with different molecular weights led to different film interconnection, grain boundary, crystal dimension and charge transport. PEO with different molecular weights affected crystal alignment, film coverage, and charge carrier mobility. Using these representative organic semiconductor and polymer additive examples, we demonstrate in this work an easy and useful pathway to manipulate the charge transport of solution-processable organic semiconductors more effectively by tuning the molecular weight of the polymer additive component.

While current research is underway in many aspects of organic electronics, we believe future endeavors that focus on

Table 1 Summary of the reviewed studies, including the authors, polymer (with its molecular weight), semiconductor, result and mobility

Authors	Polymer	Semiconductor	Result	Mobility
Leonardia <i>et al.</i>	PS (3 K, 100 K)	PDPP-TT(1)-SVS(9)	PS resulted in vertical phase segregation and formed a top encapsulation layer; thickness of the top layer depended on PS molecular weight	$0.22 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Niazi <i>et al.</i>	PS (2.2 K, 900 K)	diF-TES-ADT	PS molecular weight impacted film morphology, phase segregation and compositions	$6.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Haase <i>et al.</i>	PS (2 K, 20 K, 200 K, 2000 K)	C ₈ -BTBT	Increase in molecular weight of PS resulted in spherulitic to ribbon-like morphology	$14.4 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Singsumphan <i>et al.</i>	PS (13 K, 35 K, 280 K)	TIPS-anthracene	PS of different molecular weights induced different thin film morphologies due to difference in viscosity and processability	Not reported
Rocha <i>et al.</i>	PS (2 K, 20 K, 200 K, 2000 K)	TIPS pentacene	Different PS molecular weights impacted intermixing, phase segregation, morphology and charge transport	$12.3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Han <i>et al.</i>	PS (1.3 K, 250 K, 280 K and 290 K)	P3HT	PS with a high molecular weight resulted in a more continuous P3HT microstructure, enhancing gas sensor response	$0.03 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Ohe <i>et al.</i>	P α MS (2 K, 20 K, 60 K, 100 K, 800 K)	TIPS pentacene	P α MS with a high molecular weight resulted in a positive Gibbs free energy upon mixing and vertical phase segregation	$0.1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Kang <i>et al.</i>	P α MS (1.3 K and 575 K)	TIPS pentacene	P α MS with a high molecular weight enhanced vertical phase segregation and induced a thicker TIPS pentacene at the charge transport interface	$0.54 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
Kim <i>et al.</i>	PMMA (15 K, 120 K, 350 K, and 996 K)	C ₈ -BTBT	Crystal dimensions of C ₈ -BTBT increased with molecular weight of PMMA; thermal cycling tunes resistivity and thereby charge transport of C ₈ -BTBT	$15.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
He <i>et al.</i>	PEO (8 K, 100 K)	TIPS pentacene	PEO with 8 K molecular weight improved crystal alignment, whereas PEO with 100 K increased film coverage	$0.017 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$



these following topics will likely yield exciting discoveries. First, chemical engineering of the polymeric structures is a useful way to tune the molecular weight but at the same time add new properties to the polymers. For instance, side chain engineering can bestow solubility in organic solvents,^{211,212} regulate molecular packing,^{213,214} and modulate intermolecular interactions.^{215,216} Second, polymers with both a similar molecular weight and a similar molecular structure, but with different side chain lengths, present another way to tune molecular stacking,^{217,218} thin film morphology,^{219,220} and charge transport.^{221,222} Third, polymers can be deposited as an additional dielectric layer which helps tune the dielectric properties,^{223,224} eliminate substrate hydroxyl groups,²²⁵ reduce surface roughness,²²⁶ enable low-voltage operation,²²⁷ enhance film topography,²²⁸ control polarizability,²²⁹ and improve charge carrier mobility.^{230,231} Fourth, while current polymer mixing studies are predominately carried out on p-type organic semiconductors, the n-type counterparts are equally important to realize the fabrication of electronic devices such as complementary inverters and logic circuits.^{232,233} Mixing these polymer additives with n-type organic semiconductors can provide a more effective approach to control semiconductor crystallization, improve electron charge transport, and modulate surface passivation.^{234,235} Future research in these aforementioned areas may shed light on further improving the overall electronic device performance.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 D. Powell, E. V. Campbell, L. Flannery, J. Ogle, S. E. Soss and L. Whittaker-Brooks, Steric hindrance dependence on the spin and morphology properties of highly oriented self-doped organic small molecule thin films, *Mater. Adv.*, 2021, 2(1), 356–365.
- 2 M. Reali, P. Saini and C. Santato, Electronic and protonic transport in bio-sourced materials: a new perspective on semiconductivity, *Mater. Adv.*, 2021, 2(1), 15–31.
- 3 Q. Wang, S. Jiang, Y. Shi and Y. Li, Effect of access resistance on the experimentally measured temperature-carrier mobility dependence in highly-crystalline DNTT-based transistors, *Mater. Adv.*, 2020, 1(6), 1799–1804.
- 4 S. Bi, Q. Li, Y. Yan, K. Asare-Yeboah, T. Ma, C. Tang, Z. Ouyang, Z. He, Y. Liu and C. Jiang, Layer-dependent anisotropic frictional behavior in two-dimensional monolayer hybrid perovskite/ITO layered heterojunctions, *Phys. Chem. Chem. Phys.*, 2019, 21(5), 2540–2546.
- 5 J.-H. Tang and Y. Sun, Visible-light-driven organic transformations integrated with H₂ production on semiconductors, *Mater. Adv.*, 2020, 1(7), 2155–2162.
- 6 J. H. L. Ngai, C. C. M. Chan, C. H. Y. Ho, J. K. W. Ho, S. H. Cheung, H. Yin and S. K. So, A facile and robust approach to prepare fluorinated polymer dielectrics for probing the intrinsic transport behavior of organic semiconductors, *Mater. Adv.*, 2020, 1(4), 891–898.
- 7 Q. Liu, Y. Wang, L. Arunagiri, M. Khatib, S. Manzhos, K. Feron, S. E. Bottle, H. Haick, H. Yan, T. Michinobu and P. Sonar, Versatile nature of anthanthrone based polymers as active multifunctional semiconductors for various organic electronic devices, *Mater. Adv.*, 2020, 1(9), 3428–3438.
- 8 S. Bi, W. Zhao, Y. Sun, C. Jiang, Y. Liu, Z. He, Q. Li and J. Song, Dynamic photonic perovskite light-emitting diodes with post-treatment-enhanced crystallization as writable and wipeable inscribers, *Nanoscale Adv.*, 2021, 3(23), 6659–6668.
- 9 J. Guo, B. Yu, F. Zhu and D. Yan, Significant improvement of 2,9-DPh-DNTT organic thin-film transistors based on organic heterojunction buffer layer, *Org. Electron.*, 2021, 93, 106159.
- 10 W. Li, L. Li, Q. Sun, X. Liu, M. Kanehara, T. Nakayama, J. Jiu, K. Sakamoto and T. Minari, Direct fabrication of high-resolution and high-performance flexible electronics via surface-activation-localized electroless plating, *Chem. Eng. J.*, 2021, 416, 127644.
- 11 K. J. Thorley, M. Benford, Y. Song, S. R. Parkin, C. Risko and J. E. Anthony, What is special about silicon in functionalised organic semiconductors?, *Mater. Adv.*, 2021, 2(16), 5415–5421.
- 12 Z. He, Z. Zhang and S. Bi, Nanoparticles for organic electronics applications, *Mater. Res. Express*, 2020, 7, 012004.
- 13 Z. Yang, S. Lin, J. Liu, K. Zheng, G. Lu, B. Ye, J. Huang, Y. Zhang, Y. Ye, T. Guo and G. Chen, High performance phototransistors with organic/quantum dot composite materials channels, *Org. Electron.*, 2020, 78, 105565.
- 14 J. H. Chen, J. Anthony and D. C. Martin, Thermally induced solid-state phase transition of bis(triisopropylsilyl)ethynyl pentacene crystals, *J. Phys. Chem. B*, 2006, 110(33), 16397–16403.
- 15 D. Kneppel, F. Talnack, B. K. Boroujeni, C. Teixeira da Rocha, M. Höppner, A. Tahn, S. C. B. Mannsfeld, F. Ellinger, K. Leo and H. Kleemann, Solution-processed pseudo-vertical organic transistors based on TIPS-pentacene, *Mater. Today Energy*, 2021, 21, 100697.
- 16 Z. He, Z. Zhang, S. Bi, K. Asare-Yeboah, J. Chen and D. Li, A facile and novel route to improve TIPS pentacene based organic thin film transistor performance with elastomer, *Synth. Met.*, 2020, 262, 116337.
- 17 J. H. Chen, J. E. Anthony and D. C. Martin, Crystallographic cracking during a thermally-induced solid-state phase transition in TIPS pentacene, *Abstr. Pap. Am. Chem. Soc.*, 2005, 230, 309.
- 18 G. Schweicher, G. Liu, P. Fastré, R. Resel, M. Abbas, G. Wantz and Y. H. Geerts, Directional crystallization of



- C8-BTBT-C8 thin films in a temperature gradient, *Mater. Chem. Front.*, 2021, 5(1), 249–258.
- 19 R. Janneck, T. S. Nowack, F. De Roose, H. Ali, W. Dehaene, P. Heremans, J. Genoe and C. Rolin, Integration of highly crystalline C8-BTBT thin-films into simple logic gates and circuits, *Org. Electron.*, 2019, 67, 64–71.
 - 20 W. Zhao, J. Jie, Q. Wei, Z. Lu, R. Jia, W. Deng, X. Zhang and X. Zhang, A facile method for the growth of organic semiconductor single crystal arrays on polymer dielectric toward flexible field-effect transistors, *Adv. Funct. Mater.*, 2019, 29(32), 1902494.
 - 21 Z. Chai, S. A. Abbasi and A. A. Busnaina, Scalable directed assembly of highly crystalline 2,7-dioctyl[1]benzothieno[3,2-*b*][1]benzothiophene (C8-BTBT) films, *ACS Appl. Mater. Interfaces*, 2018, 10(21), 18123–18130.
 - 22 S. Bi, Y. Li, Z. He, Z. Ouyang, Q. Guo and C. Jiang, Self-assembly diketopyrrolopyrrole-based materials and polymer blend with enhanced crystal alignment and property for organic field-effect transistors, *Org. Electron.*, 2019, 65, 96–99.
 - 23 S. Y. Qu and H. Tian, Diketopyrrolopyrrole (DPP)-based materials for organic photovoltaics, *Chem. Commun.*, 2012, 48(25), 3039–3051.
 - 24 Y. Qiao, Y. Guo, C. Yu, F. Zhang, W. Xu, Y. Liu and D. Zhu, Diketopyrrolopyrrole-containing quinoidal small molecules for high-performance, air-stable, and solution-processable n-channel organic field-effect transistors, *J. Am. Chem. Soc.*, 2012, 134(9), 4084–4087.
 - 25 Z. Zhang, Z. He, S. Bi and K. Asare-Yeboah, Phase segregation controlled semiconductor crystallization for organic thin film transistors, *J. Sci.: Adv. Mater. Devices*, 2020, 5(2), 151–163.
 - 26 C.-H. Kim, Bias-stress effects in diF-TES-ADT field-effect transistors, *Solid-State Electron.*, 2019, 153, 23–26.
 - 27 S. Hunter, J. H. Chen and T. D. Anthopoulos, Microstructural control of charge transport in organic blend thin-film transistors, *Adv. Funct. Mater.*, 2014, 24(38), 5969–5976.
 - 28 J. Panidi, A. F. Paterson, D. Khim, Z. P. Fei, Y. Han, L. Tsetseris, G. Vourlias, P. A. Patsalas, M. Heeney and T. D. Anthopoulos, Remarkable enhancement of the hole mobility in several organic small-molecules, polymers, and small-molecule:polymer blend transistors by simple admixing of the Lewis acid p-dopant B(C₆F₅)(3), *Adv. Sci.*, 2018, 5(1), 1700290.
 - 29 K. N. Choi, K. S. Kim, K. S. Chung and H. Lee, Solvent Effect on the Electrical Properties of Triisopropylsilylethynyl (TIPS) Pentacene Organic Thin-Film Transistors, *IEEE Trans. Device Mater. Reliab.*, 2009, 9(3), 489–493.
 - 30 Z. He, Z. Zhang, K. Asare-Yeboah and S. Bi, Poly(α -methylstyrene) polymer and small-molecule semiconductor blend with reduced crystal misorientation for organic thin film transistors, *J. Mater. Sci.: Mater. Electron.*, 2019, 30, 14335–14343.
 - 31 J. H. Chen, D. C. Martin and J. E. Anthony, Morphology and molecular orientation of thin-film bis(triisopropylsilylethynyl) pentacene, *J. Mater. Res.*, 2007, 22(6), 1701–1709.
 - 32 W. Zhou, N. J. Yutronkie, B. H. Lessard and J. L. Brusso, From chemical curiosity to versatile building blocks: Unmasking the hidden potential of main-group phthalocyanines in organic field-effect transistors, *Mater. Adv.*, 2021, 2(1), 165–185.
 - 33 S. Sehlangia, S. Sharma, S. K. Sharma and C. P. Pradeep, 2,2'-(Arylenedivinylene)bis-8-hydroxyquinolines exhibiting aromatic π - π stacking interactions as solution-processable p-type organic semiconductors for high-performance organic field effect transistors, *Mater. Adv.*, 2021, 2(14), 4643–4651.
 - 34 T. Hodsden, K. J. Thorley, A. Basu, A. J. P. White, C. Wang, W. Mitchell, F. Glöcklhofer, T. D. Anthopoulos and M. Heeney, The influence of alkyl group regiochemistry and backbone fluorination on the packing and transistor performance of *N*-cyanoimine functionalised indaceno-dithiophenes, *Mater. Adv.*, 2021, 2(5), 1706–1714.
 - 35 A. R. Kirmani, E. F. Roe, C. M. Stafford and L. J. Richter, Role of the electronically-active amorphous state in low-temperature processed In₂O₃ thin-film transistors, *Mater. Adv.*, 2020, 1(2), 167–176.
 - 36 A. Rianjanu, R. Aflaha, N. I. Khamidy, M. Djamal, K. Triyana and H. S. Wasisto, Room-temperature ppb-level trimethylamine gas sensors functionalized with citric acid-doped polyvinyl acetate nanofibrous mats, *Mater. Adv.*, 2021, 2(11), 3705–3714.
 - 37 R. Malik, N. Joshi and V. K. Tomer, Advances in the designs and mechanisms of MoO₃ nanostructures for gas sensors: A holistic review, *Mater. Adv.*, 2021, 2(13), 4190–4227.
 - 38 J. M. Suh, T. H. Eom, S. H. Cho, T. Kim and H. W. Jang, Light-activated gas sensing: A perspective of integration with micro-LEDs and plasmonic nanoparticles, *Mater. Adv.*, 2021, 2(3), 827–844.
 - 39 K. Asare-Yeboah, Q. Li, C. Jiang, Z. He, S. Bi, Y. Liu and C. Liu, High performance and efficiency resonant photo-effect-transistor by near-field nano-strip-controlled organic light emitting diode gate, *J. Phys. Chem. Lett.*, 2020, 11(16), 6526–6534.
 - 40 S. Bi, Q. Li, Z. He, Q. Guo, K. Asare-Yeboah, Y. Liu and C. Jiang, Highly enhanced performance of integrated piezo photo-transistor with dual inverted OLED gate and nanowire array channel, *Nano Energy*, 2019, 66, 104101.
 - 41 J. Panidi, J. Kainth, A. F. Paterson, S. Wang, L. Tsetseris, A.-H. Emwas, M. A. McLachlan, M. Heeney and T. D. Anthopoulos, Introducing a nonvolatile N-type dopant drastically improves electron transport in polymer and small-molecule organic transistors, *Adv. Funct. Mater.*, 2019, 29(34), 1902784.
 - 42 D. Panigrahi, R. Hayakawa, K. Honma, K. Kanai and Y. Wakayama, Organic heterojunction transistors for mechanically flexible multivalued logic circuits, *Appl. Phys. Express*, 2021, 14(8), 081004.
 - 43 Z. Yang, S. Lin, J. Liu, K. Zheng, G. Lu, B. Ye, J. Huang, Y. Zhang, Y. Ye, T. Guo and G. Chen, Integration of highly crystalline C8-BTBT thin-films into simple logic gates and circuits, *Org. Electron.*, 2020, 78, 64–71.



- 44 Q. Li, C. Jiang, S. Bi, K. Asare-Yeboah, Z. He and Y. Liu, Photo-triggered logic circuits assembled on integrated illuminants and resonant nanowires, *ACS Appl. Mater. Interfaces*, 2020, **12**(24), 46501–46508.
- 45 C. Pitsalidis, N. Kalfagiannis, N. A. Hastas, P. G. Karagiannidis, C. Kapnopoulos, A. Ioakeimidis and S. Logothetidis, High performance transistors based on the controlled growth of triisopropylsilylethynyl-pentacene crystals *via* non-isotropic solvent evaporation, *RSC Adv.*, 2014, **4**(40), 20804–20813.
- 46 X. Wang, M. Yuan, X. Xiong, M. Chen, M. Qin, L. Qiu, H. Lu, G. Zhang, G. Lv and A. H. W. Choi, Process optimization for inkjet printing of triisopropylsilylethynyl pentacene with single-solvent solutions, *Thin Solid Films*, 2015, **578**, 11–19.
- 47 Z. He, Z. Zhang, S. Bi, K. Asare-Yeboah and J. Chen, Ultra-low misorientation angle in small-molecule semiconductor/polyethylene oxide blends for organic thin film transistors, *J. Polym. Res.*, 2020, **27**(3), 75.
- 48 Y.-H. Kim, Y. U. Lee, J.-I. Han, S.-M. Han and M.-K. Han, Influence of solvent on the film morphology, crystallinity and electrical characteristics of triisopropylsilyl pentacene OTFTs, *J. Electrochem. Soc.*, 2007, **154**(12), H995–H998.
- 49 D. K. Hwang, C. Fuentes-Hernandez, J. D. Berrigan, Y. N. Fang, J. Kim, W. J. Potscavage, H. Cheun, K. H. Sandhage and B. Kippelen, Solvent and polymer matrix effects on TIPS-pentacene/polymer blend organic field-effect transistors, *J. Mater. Chem.*, 2012, **22**(12), 5531–5537.
- 50 M. S. Ozorio, S. A. Camacho, N. J. A. Cordeiro, J. P. L. Duarte and N. Alves, Solvent effect on morphology and optical properties of poly(3-hexylthiophene): TIPS-pentacene blends, *J. Electron. Mater.*, 2018, **47**(2), 1353–1361.
- 51 M. Chen, B. Peng, S. Huang and P. K. L. Chan, Understanding the meniscus-guided coating parameters in organic field-effect-transistor fabrications, *Adv. Funct. Mater.*, 2020, **30**(1), 1905963.
- 52 M. Kondo, T. Kajitani, T. Uemura, Y. Noda, F. Ishiwari, Y. Shoji, T. Araki, S. Yoshimoto, T. Fukushima and T. Sekitani, Highly-ordered triptycene modifier layer based on blade coating for ultraflexible organic transistors, *Sci. Rep.*, 2019, **9**(1), 9200.
- 53 D. Wu, M. Kaplan, H. W. Ro, S. Engmann, D. A. Fischer, D. M. DeLongchamp, L. J. Richter, E. Gann, L. Thomsen, C. R. McNeill and X. Zhang, Blade coating aligned, high-performance, semiconducting-polymer transistors, *Chem. Mater.*, 2018, **30**(6), 1924–1936.
- 54 L. Ding, J. Zhao, Y. Huang, W. Tang, S. Chen and X. Guo, Flexible-blade coating of small molecule organic semiconductor for low voltage organic field effect transistor, *IEEE Electron Device Lett.*, 2017, **38**(3), 338–340.
- 55 R. Hofmockel, U. Zschieschang, U. Kraft, R. Rödel, N. H. Hansen, M. Stolte, F. Würthner, K. Takimiya, K. Kern, J. Pflaum and H. Klauk, High-mobility organic thin-film transistors based on a small-molecule semiconductor deposited in vacuum and by solution shearing, *Org. Electron.*, 2013, **14**(12), 3213–3221.
- 56 S. Galindo, A. Tamayo, F. Leonardi and M. Mas-Torrent, Control of polymorphism and morphology in solution sheared organic field-effect transistors, *Adv. Funct. Mater.*, 2017, **27**(25), 1700526.
- 57 Q. Meng, F. Zhang, Y. Zang, D. Huang, Y. Zou, J. Liu, G. Zhao, Z. Wang, D. Ji, C.-A. Di, W. Hu and D. Zhu, Solution-sheared ultrathin films for highly-sensitive ammonia detection using organic thin-film transistors, *J. Mater. Chem. C*, 2014, **2**(7), 1264–1269.
- 58 Z. Liu, H. A. Becerril, M. E. Roberts, Y. Nishi and Z. Bao, Experimental study and statistical analysis of solution-shearing processed organic transistors based on an asymmetric small-molecule semiconductor, *IEEE Trans. Electron Devices*, 2009, **56**(2), 176–185.
- 59 M. Stolte, M. Gsänger, R. Hofmockel, S.-L. Suraru and F. Würthner, Improved ambient operation of n-channel organic transistors of solution-sheared naphthalene diimide under bias stress, *Phys. Chem. Chem. Phys.*, 2012, **14**(41), 14181–14185.
- 60 S. Huang, B. Peng and P. K. L. Chan, Ambipolar organic field-effect transistors based on a dual-function, ultrathin and highly crystalline 2,9-didecyldinaphtho[2,3-*b*:2',3'-*f*]thieno[3,2-*b*]thiophene (C10-DNTT) layer, *Adv. Electron. Mater.*, 2017, **3**(12), 1700268.
- 61 K. Kim, K. Nam, X. Li, D. Y. Lee and S. H. Kim, Programmed design of highly crystalline organic semiconductor patterns with uniaxial alignment via blade coating for high-performance organic field-effect, *ACS Appl. Mater. Interfaces*, 2019, **11**(45), 42403–42411.
- 62 B. Y. Peng, S. Y. Huang, Z. W. Zhou and P. K. L. Chan, Solution-processed monolayer organic crystals for high-performance field-effect transistors and ultrasensitive gas sensors, *Adv. Funct. Mater.*, 2017, **27**(29), 1700999.
- 63 A. Bilgayan, S.-I. Cho, M. Abiko, K. Watanabe and M. Mizukami, Flexible, high mobility short-channel organic thin film transistors and logic circuits based on 4H-21DNTT, *Sci. Rep.*, 2021, **11**(1), 11710.
- 64 Y. Y. Zhao, X. Y. Fan, J. G. Feng, X. D. Wang, Y. C. Wu, B. Su and L. Jiang, Regulated dewetting for patterning organic single crystals with pure crystallographic orientation toward high performance field-effect transistors, *Adv. Funct. Mater.*, 2018, **28**(49), 1800470.
- 65 T. Shen, H. Zhou, X. Liu, Y. Fan, D. D. Mishra, Q. Fan, Z. Yang, X. Wang, M. Zhang and J. Li, Wettability Control of Interfaces for High-Performance Organic Thin-Film Transistors by Soluble Insulating Polymer Films, *ACS Omega*, 2020, **5**(19), 10891–10899.
- 66 X. Z. Wei Deng, H. Dong, J. Jie, X. Xu, J. Liu, L. He, L. Xu, W. Hu and X. Zhang, Channel-restricted meniscus self-assembly for uniformly aligned growth of single-crystal arrays of organic semiconductors, *Mater. Today*, 2019, **24**, 17–25.
- 67 J. Chang, C. Chi, J. Zhang and J. Wu, Controlled growth of large-area high-performance small-molecule organic single-crystalline transistors by slot-die coating using a mixed solvent system, *Adv. Mater.*, 2013, **25**(44), 6442–6447.



- 68 R. Wang, H.-j. Kwon, X. Tang, H. Ye, C. E. Park, J. Kim, H. Kong and S. H. Kim, Slot-die coating of sol-gel-based organic-inorganic nanohybrid dielectric layers for flexible and large-area organic thin film transistors, *Appl. Surf. Sci.*, 2020, **529**, 147198.
- 69 J. Chang, Z. Lin, J. Li, S. L. Lim, F. Wang, G. Li, J. Zhang and J. Wu, Enhanced polymer thin film transistor performance by carefully controlling the solution self-assembly and film alignment with slot die coating, *Adv. Electron. Mater.*, 2015, **1**(7), 1500036.
- 70 A. K. K. Kyaw, L. S. Lay, G. W. Peng, J. Changyun and Z. Jie, A nanogroove-guided slot-die coating technique for highly ordered polymer films and high-mobility transistors, *Chem. Commun.*, 2016, **52**(2), 358–361.
- 71 Z. Lin, X. Guo, L. Zhou, C. Zhang, J. Chang, J. Wu and J. Zhang, Solution-processed high performance organic thin film transistors enabled by roll-to-roll slot die coating technique, *Org. Electron.*, 2018, **54**, 80–88.
- 72 P. Xie, T. Liu, J. Sun, J. Jiang, Y. Yuan, Y. Gao, J. Zhou and J. Yang, Solution-processed ultra-flexible C8-BTBT organic thin-film transistors with the corrected mobility over 18 cm²/(V s), *Sci. Bull.*, 2020, **65**(10), 791–795.
- 73 W. Han and Z. Q. Lin, Learning from “Coffee Rings”: Ordered structures enabled by controlled evaporative self-assembly, *Angew. Chem., Int. Ed.*, 2012, **51**(7), 1534–1546.
- 74 A. J. Petsi and V. N. Burganos, Evaporation-induced flow in an inviscid liquid line at any contact angle, *Physical Review E*, 2006, **73**(4), 041201.
- 75 H. S. Lee, D. Kwak, W. H. Lee, J. H. Cho and K. Cho, Self-organization characteristics of soluble pentacene on wettability-controlled patterned substrate for organic field-effect transistors, *J. Phys. Chem. C*, 2010, **114**(5), 2329–2333.
- 76 Z. Wang, H. Guo, J. Li, L. Wang and G. Dong, Marangoni effect-controlled growth of oriented film for high performance C8-BTBT transistors, *Adv. Mater. Interfaces*, 2019, **6**(8), 1801736.
- 77 J. Soeda, T. Uemura, T. Okamoto, C. Mitsui, M. Yamagishi and J. Takeya, Inch-size solution-processed single-crystalline films of high-mobility organic semiconductors, *Appl. Phys. Express*, 2013, **6**(7), 076503.
- 78 K. Manoli, L. M. Dumitru, M. Y. Mulla, M. Magliulo, C. D. Franco, M. V. Santacroce, G. Scamarcio and L. Torsi, A comparative study of the gas sensing behavior in P3HT- and PBTTT-based OTFTs: The influence of film morphology and contact electrode position, *Sensors*, 2014, **14**(9), 16869–16880.
- 79 Y. Seo, J. H. Lee, J. E. Anthony, K. V. Nguyen, Y. H. Kim, H. W. Jang, S. Ko, Y. Cho and W. H. Lee, Effects of grain boundary density on the gas sensing properties of triethylsilylethynyl-anthradithiophene field-effect transistors, *Adv. Mater. Interfaces*, 2018, **5**(3), 1701399.
- 80 S. Hou, X. Zhuang, H. Fan and J. Yu, Grain boundary control of organic semiconductors via solvent vapor annealing for high-sensitivity NO₂ detection, *Sensors*, 2021, **21**(1), 226.
- 81 B. Shao, Y. Liu, X. Zhuang, S. Hou, S. Han, X. Yu and J. Yu, Crystallinity and grain boundary control of TIPS-pentacene in organic thin-film transistors for the ultra-high sensitive detection of NO₂, *J. Mater. Chem. C*, 2019, **7**(33), 10196–10202.
- 82 Z. He, Z. Zhang, K. Asare-Yeboah, S. Bi, J. Chen and D. Li, Polyferrocenylsilane semicrystalline polymer additive for solution-processed p-channel organic thin film transistors, *Polymers*, 2021, **13**(3), 402.
- 83 C. W. Sele, B. K. C. Kjellander, B. Niesen, M. J. Thornton, J. van der Putten, K. Myny, H. J. Wondergem, A. Moser, R. Resel, A. van Breemen, N. van Aerle, P. Heremans, J. E. Anthony and G. H. Gelinck, Controlled deposition of highly ordered soluble acene thin films: Effect of morphology and crystal orientation on transistor performance, *Adv. Mater.*, 2009, **21**(48), 4926–4931.
- 84 S. Yamazaki, T. Hamada, T. Nagase, S. Tokai, M. Yoshikawa, T. Kobayashi, Y. Michiwaki, S. Watase, M. Watanabe, K. Matsukawa and H. Naito, Drastic improvement in wettability of 6,13-bis(triisopropylsilylethynyl)pentacene by addition of silica nanoparticles for solution-processable organic field-effect transistors, *Appl. Phys. Express*, 2010, **3**(9), 091602.
- 85 J. A. Lim, H. S. Lee, W. H. Lee and K. Cho, Control of the morphology and structural development of solution-processed functionalized acenes for high-performance organic transistors, *Adv. Funct. Mater.*, 2009, **19**(10), 1515–1525.
- 86 S. S. Lee, C. S. Kim, E. D. Gomez, B. Purushothaman, M. F. Toney, C. Wang, A. Hexemer, J. E. Anthony and Y.-L. Loo, Controlling nucleation and crystallization in solution-processed organic semiconductors for thin-film transistors, *Adv. Mater.*, 2009, **21**(35), 3605–3609.
- 87 Z. He, K. Xiao, W. Durant, D. K. Hensley, J. E. Anthony, K. Hong, S. M. Kilbey, II, J. Chen and D. Li, Enhanced performance consistency in nanoparticle/tips pentacene-based organic thin film transistors, *Adv. Funct. Mater.*, 2011, **21**(19), 3617–3623.
- 88 D.-K. Kim, P. Vincent, J. Jang, I. M. Kang, H. Kim, P. Lang, M. Choi and J.-H. Bae, Contact line curvature-induced molecular misorientation of a surface energy patterned organic semiconductor in meniscus-guided coating, *Appl. Surf. Sci.*, 2020, **504**, 144362.
- 89 J. Chen, M. Shao, K. Xiao, Z. He, D. Li, B. S. Lokitz, D. K. Hensley, S. M. Kilbey, II, J. E. Anthony, J. K. Keum, A. J. Rondinone, W.-Y. Lee, S. Hong and Z. Bao, Conjugated polymer-mediated polymorphism of a high performance, small-molecule organic semiconductor with tuned intermolecular interactions, enhanced long-range order, and charge transport, *Chem. Mater.*, 2013, **25**(21), 4378–4386.
- 90 S. C. B. Mannsfeld, M. L. Tang and Z. Bao, Thin film structure of triisopropylsilylethynyl-functionalized pentacene and tetraceno[2,3-*b*]thiophene from grazing incidence X-ray diffraction, *Adv. Mater.*, 2011, **23**(1), 127–131.
- 91 Z. He, Z. Zhang and S. Bi, Nanoscale alignment of semiconductor crystals for high-fidelity organic electronics applications, *Appl. Nanosci.*, 2021, **11**, 787–795.



- 92 H. Yoo, H. H. Choi, T. J. Shin, T. Rim, K. Cho, S. Jung and J.-J. Kim, Self-assembled, millimeter-sized TIPS-pentacene spherulites grown on partially crosslinked polymer gate dielectric, *Adv. Funct. Mater.*, 2015, **25**(24), 3658–3665.
- 93 Z. He, Z. Zhang, S. Bi and J. Chen, Tuning charge transport in organic semiconductors with nanoparticles and hexamethyldisilazane, *J. Nanopart. Res.*, 2021, **23**(1), 5.
- 94 J. Rivnay, L. H. Jimison, J. E. Northrup, M. F. Toney, R. Noriega, S. Lu, T. J. Marks, A. Facchetti and A. Salleo, Large modulation of carrier transport by grain-boundary molecular packing and microstructure in organic thin films, *Nat. Mater.*, 2009, **8**(12), 952–958.
- 95 Z. He, Z. Zhang and S. Bi, Long-range crystal alignment with polymer additive for organic thin film transistors, *J. Polym. Res.*, 2019, **26**(7), 173.
- 96 D. Choi, B. Ahn, S. H. Kim, K. Hong, M. Ree and C. E. Park, High-performance triisopropylsilylethynyl pentacene transistors via spin coating with a crystallization-assisting layer, *ACS Appl. Mater. Interfaces*, 2012, **4**(1), 117–122.
- 97 Z. He, J. Chen, J. K. Keum, G. Szulczewski and D. Li, Improving performance of TIPS pentacene-based organic thin film transistors with small-molecule additives, *Org. Electron.*, 2014, **15**(1), 150–155.
- 98 R. Jia, X. Wu, W. Deng, X. Zhang, L. Huang, K. Niu, L. Chi and J. Jie, Unraveling the mechanism of the persistent photoconductivity in organic phototransistors, *Adv. Funct. Mater.*, 2019, **29**(45), 1905657.
- 99 J. H. Chen, C. K. Tee, M. Shtein, D. C. Martin and J. Anthony, Controlled solution deposition and systematic study of charge-transport anisotropy in single crystal and single-crystal textured TIPS pentacene thin films, *Org. Electron.*, 2009, **10**(4), 696–703.
- 100 Z. He, N. Lopez, X. Chi and D. Li, Solution-based 5,6,11,12-tetrachlorotetracene crystal growth for high-performance organic thin film transistors, *Org. Electron.*, 2015, **22**, 191–196.
- 101 S. Bi, Z. He, J. Chen and D. Li, Solution-grown small-molecule organic semiconductor with enhanced crystal alignment and areal coverage for organic thin film transistors, *AIP Adv.*, 2015, **5**(7), 077170.
- 102 J. H. Chen, C. K. Tee, J. Y. Yang, C. Shaw, M. Shtein, J. Anthony and D. C. Martin, Thermal and mechanical cracking in bis(triisopropylsilylethynyl) pentacene thin films, *J. Polym. Sci. Pol. Phys.*, 2006, **44**(24), 3631–3641.
- 103 K. Asare-Yeboah, R. M. Frazier, G. Szulczewski and D. Li, Temperature gradient approach to grow large, preferentially oriented 6,13-bis(triisopropylsilylethynyl) pentacene crystals for organic thin film transistors, *J. Vac. Sci. Technol. B*, 2014, **32**(5), 052401.
- 104 M. Y. Cho, Y. D. Han, H. S. Kang, K. Kim, K. H. Kim, M. J. Cho, D. H. Choi and J. Joo, Photoresponsive characteristics and hysteresis of soluble 6,13-bis(triisopropylsilylethynyl)-pentacene-based organic thin film transistors with and without annealing, *J. Appl. Phys.*, 2010, **107**(3), 6.
- 105 Z. He, J. Chen and D. Li, Crystal alignment for high performance organic electronics devices, *J. Vac. Sci. Technol. A*, 2019, **37**(4), 040801.
- 106 M. J. Lee, D. Gupta, N. Zhao, M. Heeney, I. McCulloch and H. Sirringhaus, Anisotropy of charge transport in a uniaxially aligned and chain-extended, high-mobility, conjugated polymer semiconductor, *Adv. Funct. Mater.*, 2011, **21**(5), 932–940.
- 107 Z. He, J. Chen, Z. Sun, G. Szulczewski and D. Li, Air-flow navigated crystal growth for TIPS pentacene-based organic thin-film transistors, *Org. Electron.*, 2012, **13**(10), 1819–1826.
- 108 M. J. Kim, H. W. Heo, Y. K. Suh and C. K. Song, Morphology control of TIPS-pentacene grains with inert gas injection and effects on the performance of OTFTs, *Org. Electron.*, 2011, **12**(7), 1170–1176.
- 109 L. Yu, X. Li, J. Smith, S. Tierney, R. Sweeney, B. K. C. Kjellander, G. H. Gelinck, T. D. Anthopoulos and N. Stingelin, Solution-processed small molecule transistors with low operating voltages and high grain-boundary anisotropy, *J. Mater. Chem.*, 2012, **22**(19), 9458–9461.
- 110 K. V. Nguyen, M. M. Payne, J. E. Anthony, J. H. Lee, E. Song, B. Kang, K. Cho and W. H. Lee, Grain boundary induced bias instability in soluble acene-based thin-film transistors, *Sci. Rep.*, 2016, **6**(1), 33224.
- 111 Z. He, K. Asare-Yeboah, Z. Zhang and S. Bi, Self-assembly crystal microribbons with nucleation additive for high-performance organic thin film transistors, *Jpn. J. Appl. Phys.*, 2019, **58**, 061009.
- 112 X. Meng, Z. Liu, B. Cui, D. Qin, H. Geng, W. Cai, L. Fu, J. He, Z. Ren and J. Sui, Grain boundary engineering for achieving high thermoelectric performance in n-type Skutterudites, *Adv. Energy Mater.*, 2017, 1602582.
- 113 Z. He, Z. Zhang, K. Asare-Yeboah, S. Bi, J. Chen and D. Li, Crystal growth of small-molecule organic semiconductors with nucleation additive, *Curr. Appl. Phys.*, 2021, **21**, 107–115.
- 114 Y. Sun, Z. Zhang, K. Asare-Yeboah, S. Bi and Z. He, Poly(butyl acrylate) polymer enhanced phase segregation and morphology of organic semiconductor for solution-processed thin film transistors, *J. Appl. Polym. Sci.*, 2021, **138**(27), 50654.
- 115 Z. R. He, J. H. Chen, J. K. Keum, G. Szulczewski and D. W. Li, Improving performance of TIPS pentacene-based organic thin film transistors with small-molecule additives, *Org. Electron.*, 2014, **15**(1), 150–155.
- 116 S. H. Jin, K. D. Jung, H. Shin, B.-G. Park and J. D. Lee, Grain size effects on contact resistance of top-contact pentacene TFTs, *Synth. Met.*, 2006, **156**(2), 196–201.
- 117 Z. He, Z. Zhang and S. Bi, Small-molecule additives for organic thin film transistors, *J. Mater. Sci.: Mater. Electron.*, 2019, **30**, 20899–20913.
- 118 J. Zhang, J. P. Rabe and N. Koch, Grain-boundary evolution in a pentacene monolayer, *Adv. Mater.*, 2008, **20**(17), 3254–3257.
- 119 M. Hirose, E. Tsunemi, K. Kobayashi and H. Yamada, Influence of grain boundary on electrical properties of organic crystalline grains investigated by dual-probe atomic force microscopy, *Appl. Phys. Lett.*, 2013, **103**(17), 173109.



- 120 M. Weis, K. Gmucová, V. Nádaždy, E. Majková, D. Haško, D. Taguchi, T. Manaka and M. Iwamoto, Grain boundary effect on charge transport in pentacene thin films, *Jpn. J. Appl. Phys.*, 2011, **50**(4), 04DK03.
- 121 A. Di Carlo, F. Piacenza, A. Bolognesi, B. Stadlober and H. Maresch, Influence of grain sizes on the mobility of organic thin-film transistors, *Appl. Phys. Lett.*, 2005, **86**(26), 263501.
- 122 J. Levinson, F. R. Shepherd, P. J. Scanlon, W. D. Westwood, G. Este and M. Rider, Conductivity behavior in polycrystalline semiconductor thin film transistors, *J. Appl. Phys.*, 1982, **53**(2), 1193–1202.
- 123 F. V. Farmakis, J. Brini, G. Kamarinos, C. T. Angelis, C. A. Dimitriadis and M. Miyasaka, On-current modeling of large-grain polycrystalline silicon thin-film transistors, *IEEE Trans. Electron Devices*, 2001, **48**(4), 701–706.
- 124 Z. He, J. Chen and D. Li, Polymer additive controlled morphology for high performance organic thin film transistors, *Soft Matter*, 2019, **15**(29), 5790–5803.
- 125 H. Chung and Y. Diao, Polymorphism as an emerging design strategy for high performance organic electronics, *J. Mater. Chem. C*, 2016, **4**(18), 3915–3933.
- 126 T. Shen, H. Zhou, J. Xin, Q. Fan, Z. Yang, J. Wang, T. Mei, X. Wang, N. Wang and J. Li, Controllable microstructure of polymer-small molecule blend thin films for high-performance organic field-effect transistors, *Appl. Surf. Sci.*, 2019, **498**, 143822.
- 127 Y. Sun, Z. Zhang, K. Asare-Yeboah, S. Bi and Z. He, Large-dimensional organic semiconductor crystals with poly(butyl acrylate) polymer for solution-processed organic thin film transistors, *Electron Mater. Lett.*, 2021, **17**, 33–42.
- 128 J. Smith, R. Hamilton, I. McCulloch, M. Heeney, J. E. Anthony, D. D. C. Bradley and T. D. Anthopoulos, High mobility p-channel organic field effect transistors on flexible substrates using a polymer-small molecule blend, *Synth. Met.*, 2009, **159**(21–22), 2365–2367.
- 129 X. Xiao, G. Pan and F. Zhang, High transparency and enhanced mobility of field-effect transistors of the semiconductor/insulator polymer blends with ultralow semiconductor content, *Org. Electron.*, 2020, **82**, 105709.
- 130 M. Nikolka, I. Nasrallah, B. Rose, M. K. Ravva, K. Broch, A. Sadhanala, D. Harkin, J. Charmet, M. Hurhangee, A. Brown, S. Illig, P. Too, J. Jongman, I. McCulloch, J. L. Bredas and H. Sirringhaus, High operational and environmental stability of high-mobility conjugated polymer field-effect transistors through the use of molecular additives, *Nat. Mater.*, 2017, **16**(3), 356–362.
- 131 K. Fukuda, Y. Takeda, M. Mizukami, D. Kumaki and S. Tokito, Fully solution-processed flexible organic thin film transistor arrays with high mobility and exceptional uniformity, *Sci. Rep.*, 2014, **4**(1), 3947.
- 132 Y. H. Kim, J. E. Anthony and S. K. Park, Polymer blended small molecule organic field effect transistors with improved device-to-device uniformity and operational stability, *Org. Electron.*, 2012, **13**(7), 1152–1157.
- 133 Y. Li, C. Liu, A. Kumatani, P. Darmawan, T. Minari and K. Tsukagoshi, Large plate-like organic crystals from direct spin-coating for solution-processed field-effect transistor arrays with high uniformity, *Org. Electron.*, 2012, **13**(2), 264–272.
- 134 R. Hamilton, J. Smith, S. Ogier, M. Heeney, J. E. Anthony, I. McCulloch, J. Veres, D. D. C. Bradley and T. D. Anthopoulos, High-performance polymer-small molecule blend organic transistors, *Adv. Mater.*, 2009, **21**(10–11), 1166–1171.
- 135 F. Aikawa, J. Ueno, T. Kashiwagi and E. Itoh, Improvement of field-effect transistor performance with highly oriented, vertically phase separated TIPS-pentacene/polystyrene blends on high-k metal oxide films by using meniscus coating, *Jpn. J. Appl. Phys.*, 2019, **59**, SCCA10.
- 136 N. Onojima, Y. Mori, T. Ozawa, T. Sugai, N. Akiyama and S. Obata, Flexible organic field-effect transistors based on 6,13-bis(triisopropylsilyl)ethynyl pentacene/polystyrene blend film prepared by electrostatic spray deposition, *Jpn. J. Appl. Phys.*, 2019, **59**, SDDA13.
- 137 F. Leonardi, Q. Zhang, Y.-H. Kim and M. Mas-Torrent, Solution-sheared thin films of a donor-acceptor random copolymer/polystyrene blend as active material in field-effect transistors, *Mater. Sci. Semicond. Process.*, 2019, **93**, 105–110.
- 138 N. Onojima, K. Hara and A. Nakamura, Vertical phase separation of 6,13-bis(triisopropylsilyl)ethynyl pentacene/poly(methyl methacrylate) blends prepared by electrostatic spray deposition for organic field-effect transistors, *Jpn. J. Appl. Phys.*, 2017, **56**, 5S2.
- 139 Y. J. Kwon, Y. D. Park and W. H. Lee, Inkjet-printed organic transistors based on organic semiconductor/insulating polymer blends, *Materials*, 2016, **9**(8), 650.
- 140 J. H. Park, Y. T. Lee, H. S. Lee, J. Y. Lee, K. Lee, G. B. Lee, J. Han, T. W. Kim and S. Im, Origin of bias-stress induced instability in organic thin-film transistors with semiconducting small-molecule/insulating polymer blend channel, *ACS Appl. Mater. Interfaces*, 2013, **5**(5), 1625–1629.
- 141 H. L. Zhong, J. Smith, S. Rossbauer, A. J. P. White, T. D. Anthopoulos and M. Heeney, Air-stable and high-mobility n-channel organic transistors based on small-molecule/polymer semiconducting blends, *Adv. Mater.*, 2012, **24**(24), 3205–3211.
- 142 X. Guo, R. P. Ortiz, Y. Zheng, Y. Hu, Y.-Y. Noh, K.-J. Baeg, A. Facchetti and T. J. Marks, Bithiophene-imide-based polymeric semiconductors for field-effect transistors: Synthesis, structure–property correlations, charge carrier polarity, and device stability, *J. Am. Chem. Soc.*, 2011, **133**(5), 1405–1418.
- 143 T. Endo, T. Nagase, T. Kobayashi, K. Takimiya, M. Ikeda and H. Naito, Solution-processed dioctylbenzothienobenzothiophene-based top-gate organic transistors with high mobility, low threshold voltage, and high electrical stability, *Appl. Phys. Express*, 2010, **3**(12), 121601.
- 144 S. Georgakopoulos, D. Sparrowe, F. Meyer and M. Shkunov, Stability of top- and bottom-gate amorphous polymer field-effect transistors, *Appl. Phys. Lett.*, 2010, **97**(24), 243507.



- 145 K. Sim, Y. Choi, H. Kim, S. Cho, S. C. Yoon and S. Pyo, Soluble pentacene thin-film transistor using a high solvent and heat resistive polymeric dielectric with low-temperature processability and its long-term stability, *Org. Electron.*, 2009, **10**(3), 506–510.
- 146 J. A. Lim, J. H. Kim, L. Qiu, W. H. Lee, H. S. Lee, D. Kwak and K. Cho, Inkjet-printed single-droplet organic transistors based on semiconductor nanowires embedded in insulating polymers, *Adv. Funct. Mater.*, 2010, **20**(19), 3292–3297.
- 147 F. C. Chen and C. H. Liao, Improved air stability of n-channel organic thin-film transistors with surface modification on gate dielectrics, *Appl. Phys. Lett.*, 2008, **93**(10), 103310.
- 148 J. Soeda, T. Okamoto, C. Mitsui and J. Takeya, Stable growth of large-area single crystalline thin films from an organic semiconductor/polymer blend solution for high-mobility organic field-effect transistors, *Org. Electron.*, 2016, **39**, 127–132.
- 149 J. Smith, W. M. Zhang, R. Sougrat, K. Zhao, R. P. Li, D. K. Cha, A. Amassian, M. Heeney, I. McCulloch and T. D. Anthopoulos, Solution-processed small molecule-polymer blend organic thin-film transistors with hole mobility greater than 5 cm²/Vs, *Adv. Mater.*, 2012, **24**(18), 2441–2446.
- 150 J. A. Lim, W. H. Lee, H. S. Lee, J. H. Lee, Y. D. Park and K. Cho, Self-organization of ink-jet-printed triisopropylsilyl ethynyl pentacene via evaporation-induced flows in a drying droplet, *Adv. Funct. Mater.*, 2008, **18**(2), 229–234.
- 151 H. B. Akkerman, H. Y. Li and Z. N. Bao, TIPS-pentacene crystalline thin film growth, *Org. Electron.*, 2012, **13**(10), 2056–2062.
- 152 D. T. James, B. K. C. Kjellander, W. T. T. Smaal, G. H. Gelinck, C. Combe, I. McCulloch, R. Wilson, J. H. Burroughes, D. D. C. Bradley and J. S. Kim, Thin-film morphology of inkjet-printed single-droplet organic transistors using polarized raman spectroscopy: Effect of blending TIPS-pentacene with insulating polymer, *Acs Nano*, 2011, **5**(12), 9824–9835.
- 153 J. I. Han, C.-Y. Lim, S. K. Park and Y.-H. Kim, High-performance 2,8-difluoro-5,11-bis(triethylsilyl ethynyl) anthradithiophene thin-film transistors facilitated by pre-deposited ink-jet blending, *Jpn. J. Appl. Phys.*, 2013, **52**(3R), 031601.
- 154 X. Lei, S. Ge, Y. Tan, Z. Wang, J. Li, X. Li, G. Hu, X. Zhu, M. Huang, Y. Zhu and B. Xiang, High capacity and energy density of Zn–Ni–Co–P nanowire arrays as an advanced electrode for aqueous asymmetric supercapacitor, *ACS Appl. Mater. Interfaces*, 2020, **12**(8), 9158–9168.
- 155 S. Hou, J. Yu, X. Zhuang, D. Li, Y. Liu, Z. Gao, T. Sun, F. Wang and X. Yu, Phase separation of P3HT/PMMA blend film formed semiconducting and dielectric layers in organic thin film transistors for high sensitivity NO₂ detection, *ACS Appl. Electron. Mater.*, 2019, **11**(47), 44521–44527.
- 156 N. Onojima, S. Obata, A. Nakamura and K. Hara, Influence of phase-separated morphology on small molecule/polymer blend organic field-effect transistors fabricated using electrostatic spray deposition, *Thin Solid Films*, 2017, **640**, 99–103.
- 157 S. G. Lee, H. S. Lee, S. Lee, C. W. Kim and W. H. Lee, Thickness-dependent electrical properties of soluble acene–polymer blend semiconductors, *Org. Electron.*, 2015, **24**, 113–119.
- 158 X. Xu, T. Xiao, X. Gu, X. Yang, S. V. Kershaw, N. Zhao, J. Xu and Q. Miao, Solution-processed ambipolar organic thin-film transistors by blending p- and n-type semiconductors: solid solution *versus* microphase separation, *ACS Appl. Mater. Interfaces*, 2015, **7**(51), 28019–28026.
- 159 Z. He, D. Li, D. K. Hensley, A. J. Rondinone and J. Chen, Switching phase separation mode by varying the hydrophobicity of polymer additives in solution-processed semiconducting small-molecule/polymer blends, *Appl. Phys. Lett.*, 2013, **103**(11), 113301.
- 160 W. H. Lee, D. Kwak, J. E. Anthony, H. S. Lee, H. H. Choi, D. H. Kim, S. G. Lee and K. Cho, The influence of the solvent evaporation rate on the phase separation and electrical performances of soluble acene–polymer blend semiconductors, *Adv. Funct. Mater.*, 2012, **22**(2), 267–281.
- 161 A. C. Arias, F. Endicott and R. A. Street, Surface-induced self-encapsulation of polymer thin-film transistors, *Adv. Mater.*, 2006, **18**(21), 2900–2904.
- 162 C. Liu, Y. Li, M. V. Lee, A. Kumatani and K. Tsukagoshi, Self-assembly of semiconductor/insulator interfaces in one-step spin-coating: A versatile approach for organic field-effect transistors, *Phys. Chem. Chem. Phys.*, 2013, **15**(21), 7917–7933.
- 163 Z. He, Z. Zhang, S. Bi, J. Chen and D. Li, Conjugated polymer controlled morphology and charge transport of small-molecule organic semiconductors, *Sci. Rep.*, 2020, **10**(1), 4344.
- 164 H. Shin, J. Park and J. S. Choi, P-14: Electrical characteristics of P3HT:TIPS-pentacene blend organic thin-film transistor under light irradiation, *SID Symp. Dig. Tech. Pap.*, 2020, **51**(1), 1362–1364.
- 165 F. Shiono, H. Abe, T. Nagase, T. Kobayashi and H. Naito, Optical memory characteristics of solution-processed organic transistors with self-organized organic floating gates for printable multi-level storage devices, *Org. Electron.*, 2019, **67**, 109–115.
- 166 C. M. Benavides, M. Biele, O. Schmidt, C. J. Brabec and S. F. Tedde, TIPS pentacene as a beneficial interlayer for organic photodetectors in imaging applications, *IEEE Trans. Electron Devices*, 2018, **65**(4), 1516–1522.
- 167 M. D. Ozorio, G. L. Nogueira, R. M. Morais, C. D. Martin, C. J. L. Constantino and N. Alves, Poly(3-hexylthiophene): TIPS-pentacene blends aiming transistor applications, *Thin Solid Films*, 2016, **608**, 97–101.
- 168 S. Mansouri, L. El Mir, A. A. Al-Ghamdi, O. A. Al-Hartomy, S. A. F. Al Said and F. Yakuphanoglu, Characterization and modeling of TIPS-pentacene-poly(3-hexyl) thiophene blend organic thin film transistor, *Synth. Met.*, 2013, **185**, 153–158.



- 169 R. K. Layek, V. S. Parihar, M. Skrifvars, F. Javanshour, M. Kroon, M. Kanerva, J. Vuorinen, M. Kellomäki and E. Sarlin, Tailoring of the physical and mechanical properties of biocompatible graphene oxide/gelatin composite nanolaminates *via* altering the crystal structure and morphology, *Mater. Adv.*, 2021, **2**(14), 4781–4791.
- 170 P. Bhagabati, D. Das and V. Katiyar, Bamboo-flour-filled cost-effective poly(ϵ -caprolactone) biocomposites: A potential contender for flexible cryo-packaging applications, *Mater. Adv.*, 2021, **2**(1), 280–291.
- 171 K. Wu, G. Wu, A. J. MacRobert, E. Allan, A. Gavrilidis and I. P. Parkin, Ultra high molecular weight polyethylene with incorporated crystal violet and gold nanoclusters is antimicrobial in low intensity light and in the dark, *Mater. Adv.*, 2020, **1**(9), 3339–3348.
- 172 M. Ito, Y. Yamashita, T. Mori, K. Ariga, J. Takeya and S. Watanabe, Band mobility exceeding $10 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ assessed by field-effect and chemical double doping in semicrystalline polymeric semiconductors, *Appl. Phys. Lett.*, 2021, **119**(1), 013302.
- 173 C. Wang, W. Y. Lee, D. S. Kong, R. Pfattner, G. Schweicher, R. Nakajima, C. Lu, J. G. Mei, T. H. Lee, H. C. Wu, J. Lopez, Y. Diao, X. D. Gu, S. Himmelberger, W. J. Niu, J. R. Matthews, M. Q. He, A. Salleo, Y. Nishi and Z. N. Bao, Significance of the double-layer capacitor effect in polar rubbery dielectrics and exceptionally stable low-voltage high transconductance organic transistors, *Sci. Rep.*, 2015, **5**, 17849.
- 174 B. Burkhart, P. P. Khlyabich and B. C. Thompson, Influence of the ethylhexyl side-chain content on the open-circuit voltage in rr-Poly(3-hexylthiophene-co-3-(2-ethylhexyl)thiophene) copolymers, *Macromolecules*, 2012, **45**(9), 3740–3748.
- 175 E. Orgiu, A. M. Masillamani, J.-O. Vogel, E. Treossi, A. Kiersnowski, M. Kastler, W. Pisula, F. Dötzt, V. Palermo and P. Samori, Enhanced mobility in P3HT-based OTFTs upon blending with a phenylene–thiophene–thiophene–phenylene small molecule, *Chem. Commun.*, 2012, **48**(10), 1562–1564.
- 176 N. Onojima, N. Akiyama, Y. Mori, T. Sugai and S. Obata, Small molecule/polymer blends prepared by environmentally-friendly process for mechanically-stable flexible organic field-effect transistors, *Org. Electron.*, 2020, **78**, 105597.
- 177 Y. Wang, D. Hu, M. Chen, X. Wang, H. Lu, G. Zhang, X. Wang, Z. Wu and L. Qiu, Modulation of surface solubility and wettability for high-performance inkjet-printed organic transistors, *Org. Electron.*, 2014, **15**(11), 3101–3110.
- 178 P. S. Jo, D. T. Duong, J. Park, R. Sinclair and A. Salleo, Control of rubrene polymorphs *via* polymer binders: Applications in organic field-effect transistors, *Chem. Mater.*, 2015, **27**(11), 3979–3987.
- 179 C. He, Y. He, X. Liu, A. Li, J. Chen and H. Meng, Enhancing the performance of solution-processed organic thin-film transistors by blending binary compatible small molecule semiconductors, *Org. Electron.*, 2019, **64**, 104–109.
- 180 I. Temiño, F. G. Del Pozo, M. R. Ajayakumar, S. Galindo, J. Puigdollers and M. Mas-Torrent, A rapid, low-cost, and scalable technique for printing state-of-the-art organic field-effect transistors, *Adv. Mater. Technol.*, 2016, **1**(5), 1600090.
- 181 F. Leonardi, Q. Zhang, Y.-H. Kim and M. Mas-Torrent, Solution-sheared thin films of a donor-acceptor random copolymer/polystyrene blend as active material in field-effect transistors, *Mater. Sci. Semicond. Process.*, 2019, **93**, 105–110.
- 182 M. R. Niazi, R. P. Li, E. Q. Li, A. R. Kirmani, M. Abdelsamie, Q. X. Wang, W. Y. Pan, M. M. Payne, J. E. Anthony, D. M. Smilgies, S. T. Thoroddsen, E. P. Giannelis and A. Amassian, Solution-printed organic semiconductor blends exhibiting transport properties on par with single crystals, *Nat. Commun.*, 2015, **6**, 8598.
- 183 K. Haase, C. T. da Rocha, C. Hauenstein, Y. C. Zheng, M. Hambsch and S. C. B. Mannsfeld, High-mobility, solution-processed organic field-effect transistors from C8-BTBT:polystyrene blends, *Adv. Electron. Mater.*, 2018, **4**(8), 1800076.
- 184 K. Singsumphan, Influence of polymer on morphological and optical properties of small molecules/polymer blend films, *Mater. Today Proc.*, 2017, **4**(5), 6478–6484.
- 185 C. T. da Rocha, K. Haase, Y. C. Zheng, M. Löffler, M. Hambsch and S. C. B. Mannsfeld, Solution coating of small molecule/polymer blends enabling ultralow voltage and high-mobility organic transistors, *Adv. Electron. Mater.*, 2018, **4**(8), 1800141.
- 186 S. Han, X. Zhuang, W. Shi, X. Yang, L. Li and J. Yu, Poly(3-hexylthiophene)/polystyrene (P3HT/PS) blends based organic field-effect transistor ammonia gas sensor, *Sens. Actuators, B*, 2016, **225**, 10–15.
- 187 G. D. Jones, R. E. Friedrich, T. E. Werkema and R. L. Zimmerman, Poly-alpha-methylstyrene, *Ind. Eng. Chem.*, 1956, **48**(12), 2123–2131.
- 188 J. H. Lee, H. H. Choi, Y. D. Park, J. E. Anthony, J. A. Lim, J. Cho, D. S. Chung, J. Hwang, H. W. Jang, K. Cho and W. H. Lee, 1D *versus* 2D growth of soluble acene crystals from soluble acene/polymer blends governed by a residual solvent reservoir in a phase-separated polymer matrix, *Adv. Funct. Mater.*, 2018, **28**(34), 1802875.
- 189 L.-H. Chou, W.-C. Chang, G.-Y. He, Y.-C. Chiu and C.-L. Liu, Controllable electrical performance of spray-coated semiconducting small molecule/insulating polymer blend thin film for organic field effect transistors application, *React. Funct. Polym.*, 2016, **108**, 130–136.
- 190 K. Sim, A. K. Palai, G. Tarsoly, H. Na and S. Pyo, Polymer binder assisted, solution processed cyanophenyl functionalized diketopyrrolopyrrole microwire for n-channel field-effect transistors, *Synth. Met.*, 2019, **250**, 152–160.
- 191 K. Asare-Yeboah, S. Bi, Z. He and D. Li, Temperature gradient controlled crystal growth from TIPS pentacene-poly(alpha-methyl styrene) blends for improving performance of organic thin film transistors, *Org. Electron.*, 2016, **32**, 195–199.



- 192 N. L. Vaklev, R. Muller, B. V. O. Muir, D. T. James, R. Pretot, P. van der Schaaf, J. Genoe, J. S. Kim, J. H. G. Steinke and A. J. Campbell, High-performance flexible bottom-gate organic field-effect transistors with gravure printed thin organic dielectric, *Adv. Mater. Interfaces*, 2014, **1**(3), 1300123.
- 193 W. C. Xu, H. X. He, X. S. Jing, S. J. Wu, Z. Zhang, J. W. Gao, X. S. Gao, G. F. Zhou, X. B. Lu and J. M. Liu, High performance organic nonvolatile memory transistors based on HfO₂ and poly(α -methylstyrene) electret hybrid charge-trapping layers, *Appl. Phys. Lett.*, 2017, **111**(6), 063302.
- 194 Y. S. Chung, N. Shin, J. Kang, Y. Jo, V. M. Prabhu, S. K. Satija, R. J. Kline, D. M. DeLongchamp, M. F. Toney, M. A. Loth, B. Purushothaman, J. E. Anthony and D. Y. Yoon, Zone-refinement effect in small molecule-polymer blend semiconductors for organic thin-film transistors, *J. Am. Chem. Soc.*, 2011, **133**(3), 412–415.
- 195 Z. He, S. Shaik, S. Bi, J. Chen and D. Li, Air-stable solution-processed n-channel organic thin film transistors with polymer-enhanced morphology, *Appl. Phys. Lett.*, 2015, **106**(18), 183301.
- 196 T. Ohe, M. Kuribayashi, A. Tsuboi, K. Satori, M. Itabashi and K. Nomoto, Organic thin-film transistors with phase separation of polymer-blend small-molecule semiconductors: Dependence on molecular weight and types of polymer, *Appl. Phys. Express*, 2009, **2**(12), 121502.
- 197 J. Kang, N. Shin, D. Y. Jang, V. M. Prabhu and D. Y. Yoon, Structure and properties of small molecule-polymer blend semiconductors for organic thin film transistors, *J. Am. Chem. Soc.*, 2008, **130**(37), 12273–12275.
- 198 U. Ali, K. J. B. A. Karim and N. A. Buang, A review of the properties and applications of poly(methyl methacrylate) (PMMA), *Polym. Rev.*, 2015, **55**(4), 678–705.
- 199 H. Li, Z. He, Z. Ouyang, S. Palchoudhury, C. W. Ingram, I. I. Harruna and D. Li, Modifying electrical and magnetic properties of single-walled carbon nanotubes by decorating with iron oxide nanoparticles, *J. Nanosci. Nanotechnol.*, 2020, **20**(4), 2611–2616.
- 200 C. Liu, T. Minari, Y. Li, A. Kumatani, M. V. Lee, S. H. A. Pan, K. Takimiya and K. Tsukagoshi, Direct formation of organic semiconducting single crystals by solvent vapor annealing on a polymer base film, *J. Mater. Chem.*, 2012, **22**(17), 8462–8469.
- 201 X. Zhu, Q. Wang, X. Tian, X. Zhang, Y. Feng, W. Feng, R. Li and W. Hu, Unidirectional and crystalline organic semiconductor microwire arrays by solvent vapor annealing with PMMA as the assisting layer, *J. Mater. Chem. C*, 2018, **6**(46), 12479–12483.
- 202 L. Zhang, D. Yang, Y. Wang, H. Wang, T. Song, C. Fu, S. Yang, J. Wei, R. Liu and B. Zou, Performance enhancement of FET-based photodetector by blending P3HT with PMMA, *IEEE Photon. Technol. Lett.*, 2015, **27**(14), 1535–1538.
- 203 N. Onojima, N. Akiyama, Y. Mori, T. Sugai and S. Obata, Small molecule/polymer blends prepared by environmentally-friendly process for mechanically-stable flexible organic field-effect transistors, *Org. Electron.*, 2020, **78**(105597), 105597.
- 204 S. Y. Yeo, S. Park, Y. J. Yi, D. H. Kim and J. A. Lim, Highly sensitive flexible pressure sensors based on printed organic transistors with centro-apically self-organized organic semiconductor microstructures, *ACS Appl. Mater. Interfaces*, 2017, **9**(49), 42996–43003.
- 205 S. Jung, M. Albariqi, G. Gruntz, T. Al-Hathal, A. Peinado, E. Garcia-Caurel, Y. Nicolas, T. Toupance, Y. Bonnassieux and G. Horowitz, A TIPS-TPDO-tetraCN-based n-type organic field-effect transistor with a cross-linked pmma polymer gate dielectric, *ACS Appl. Mater. Interfaces*, 2016, **8**(23), 14701–14708.
- 206 J. Kang, J. Kim, J.-W. Jo, J. S. Heo, M.-G. Kim, Y.-H. Kim, J. Kim and S. K. Park, Photochemical molecular tailoring for efficient diffusion and reorganization of organic nanocrystals for ultra-flexible organic semiconductor arrays, *Small*, 2017, **13**(1), 1602467.
- 207 Q. Xie, L. Wang, Y. Zhu, Q. Sun and L. Wang, Highly sensitive NO₂ sensors based on organic field effect transistors with Al₂O₃/PMMA bilayer dielectrics by sol-spin coating, *Org. Electron.*, 2019, **74**, 69–76.
- 208 M.-W. Kim, S. Kwon, J. Kim, C. Lee, I. Park, J. H. Shim, I.-S. Jeong, Y.-R. Jo, B. Park, J.-H. Lee, K. Lee and B.-J. Kim, Reversible polymorphic transition and hysteresis-driven phase selectivity in single-crystalline C8-BTBT rods, *Small*, 2020, **16**(3), 1906109.
- 209 S. B. Upadhye and A. R. Rajabi-Siahboomi, in *Melt Extrusion: Materials, Technology and Drug Product Design*, ed. M. A. Repka, N. Langley and J. DiNunzio, Springer New York, New York, NY, 2013, pp. 145–158.
- 210 Z. He, Z. Zhang, S. Bi and J. Chen, Effect of polymer molecular weight on morphology and charge transport of small-molecular organic semiconductors, *Electron Mater. Lett.*, 2020, **16**(5), 441–450.
- 211 S. K. Park, J. E. Anthony and T. N. Jackson, Solution-processed TIPS-pentacene organic thin-film-transistor circuits, *IEEE Electron Device Lett.*, 2007, **28**(10), 877–879.
- 212 J. G. Park, R. Vasic, J. S. Brooks and J. E. Anthony, Characterization of functionalized pentacene field-effect transistors and its logic gate application, *J. Appl. Phys.*, 2006, **100**(4), 044511.
- 213 Z. Chen, P. Müller and T. M. Swager, Syntheses of soluble, π -stacking tetracene derivatives, *Org. Lett.*, 2006, **8**(2), 273–276.
- 214 J. E. Anthony, Functionalized acenes and heteroacenes for organic electronics, *Chem. Rev.*, 2006, **106**(12), 5028–5048.
- 215 S. Saito, K. Nakakura and S. Yamaguchi, Macrocyclic restriction with flexible alkylene linkers: A simple strategy to control the solid-state properties of π -conjugated systems, *Angew. Chem., Int. Ed.*, 2012, **51**(3), 714–717.
- 216 L. Ding, H.-B. Li, T. Lei, H.-Z. Ying, R.-B. Wang, Y. Zhou, Z.-M. Su and J. Pei, Alkylene-chain effect on microwire growth and crystal packing of π -moieties, *Chem. Mater.*, 2012, **24**(10), 1944–1949.



- 217 G. E. Park, J. Shin, D. H. Lee, M. J. Cho and D. H. Choi, Effect of branched alkyl side chains on the performance of thin-film transistors and photovoltaic cells fabricated with isoindigo-based conjugated polymers, *J. Polym. Sci., Part A: Polym. Chem.*, 2015, **53**(10), 1226–1234.
- 218 Q. Wei, S. Miyanishi, K. Tajima and K. Hashimoto, Enhanced charge transport in polymer thin-film transistors prepared by contact film transfer method, *ACS Appl. Mater. Interfaces*, 2009, **1**(11), 2660–2666.
- 219 Z. He, S. Bi, K. Asare-Yeboah and Z. Zhang, Phase segregation effect on TIPS pentacene crystallization and morphology for organic thin film transistors, *J. Mater. Sci.: Mater. Electron.*, 2020, **31**(6), 4503–4510.
- 220 Z. He, Z. Zhang and S. Bi, Polyacrylate polymer assisted crystallization: Improved charge transport and performance consistency for solution-processable small-molecule semiconductor based organic thin film transistors, *J. Sci.: Adv. Mater. Devices*, 2019, **4**(3), 467–472.
- 221 Z. Hu, B. Fu, A. Aiyar and E. Reichmanis, Synthesis and characterization of graft polymethacrylates containing conducting diphenyldithiophene for organic thin-film transistors, *J. Polym. Sci., Part A: Polym. Chem.*, 2012, **50**(2), 199–206.
- 222 S. Chen, Y. Meng, Y. Li, B. Qu and D. Zhuo, Effect of the length and branching point of alkyl side chains on DPP-thieno [3,2-*b*]thiophene copolymers for organic thin-film transistors, *Opt. Mater.*, 2019, **88**, 500–507.
- 223 S. H. Kim, S. Y. Yang, K. Shin, H. Jeon, J. W. Lee, K. P. Hong and C. E. Park, Low-operating-voltage pentacene field-effect transistor with a high-dielectric-constant polymeric gate dielectric, *Appl. Phys. Lett.*, 2006, **89**(18), 183516.
- 224 S. C. Lim, S. H. Kim, J. B. Koo, J. H. Lee, C. H. Ku, Y. S. Yang and T. Zyung, Hysteresis of pentacene thin-film transistors and inverters with cross-linked poly(4-vinylphenol) gate dielectrics, *Appl. Phys. Lett.*, 2007, **90**(17), 173512.
- 225 Y.-H. Chou, Y.-C. Chiu and W.-C. Chen, High-*k* polymer-graphene oxide dielectrics for low-voltage flexible nonvolatile transistor memory devices, *Chem. Commun.*, 2014, **50**(24), 3217–3219.
- 226 Y. Jang, D. H. Kim, Y. D. Park, J. H. Cho, M. Hwang and K. Cho, Influence of the dielectric constant of a polyvinyl phenol insulator on the field-effect mobility of a pentacene-based thin-film transistor, *Appl. Phys. Lett.*, 2005, **87**(15), 152105.
- 227 M. J. Panzer, C. R. Newman and C. D. Frisbie, Low-voltage operation of a pentacene field-effect transistor with a polymer electrolyte gate dielectric, *Appl. Phys. Lett.*, 2005, **86**(10), 103503.
- 228 B. Stadlober, M. Zirkl, M. Beutl, G. Leising, S. Bauer-Gogonea and S. Bauer, High-mobility pentacene organic field-effect transistors with a high-dielectric-constant fluorinated polymer film gate dielectric, *Appl. Phys. Lett.*, 2005, **86**(24), 242902.
- 229 A. K. Mahato, D. Bharti, P. Saxena, V. Raghuvanshi, I. Varun and S. P. Tiwari, Influence of molecular weight of polymer dielectric on the photo-response of solution-processed OFETs, *Polymer*, 2021, **224**, 123724.
- 230 D. K. Hwang, M. S. Oh, J. M. Hwang, J. H. Kim and S. Im, Hysteresis mechanisms of pentacene thin-film transistors with polymer/oxide bilayer gate dielectrics, *Appl. Phys. Lett.*, 2008, **92**(1), 013304.
- 231 X. G. Yu, N. J. Zhou, S. J. Han, H. Lin, D. B. Buchholz, J. S. Yu, R. P. H. Chang, T. J. Marks and A. Facchetti, Flexible spray-coated TIPS-pentacene organic thin-film transistors as ammonia gas sensors, *J. Mater. Chem. C*, 2013, **1**(40), 6532–6535.
- 232 H. J. Cho, D.-H. Lee, E.-K. Park, M. S. Kim, S. Y. Lee, K. Park, H. Choe, J.-H. Jeon and Y.-S. Kim, Solution-processed organic-inorganic hybrid gate insulator for complementary thin film transistor logic circuits, *Thin Solid Films*, 2019, **673**, 14–18.
- 233 J. W. Borchert, B. Y. Peng, F. Letzkus, J. N. Burghartz, P. K. L. Chan, K. Zojer, S. Ludwigs and H. Klauk, Small contact resistance and high-frequency operation of flexible low-voltage inverted coplanar organic transistors, *Nat. Commun.*, 2019, **10**, 1119.
- 234 D. Kumaki, S. Ando, S. Shimono, Y. Yamashita, T. Umeda and S. Tokito, Significant improvement of electron mobility in organic thin-film transistors based on thiazolothiazole derivative by employing self-assembled monolayer, *Appl. Phys. Lett.*, 2007, **90**(5), 053506.
- 235 L. L. Chua, J. Zaumseil, J. F. Chang, E. C. W. Ou, P. K. H. Ho, H. Sirringhaus and R. H. Friend, General observation of n-type field-effect behaviour in organic semiconductors, *Nature*, 2005, **434**(7030), 194–199.

