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# Optical transitions and radiative properties of green emitting $\text{Ho}^{3+}:\text{YVO}_4$ phosphor

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The  $\text{Ho}^{3+}$ -doped  $\text{YVO}_4$  phosphors were successfully prepared via a sol-gel process in which citric acid was used as a chelating agent. X-ray diffraction (XRD) confirmed the effective inclusion of  $\text{Ho}^{3+}$  ions into the host matrix with the formation of single phase  $\text{YVO}_4$ . The surface morphology was observed using SEM, the results of which showed a grain growth propensity and the agglomeration of prepared phosphors. The V–O ( $\text{VO}_4^{3-}$ ) vibration mode was analyzed through Fourier transform infrared (FTIR) spectra. The spectroscopic properties were reported through UV-vis-NIR diffuse reflectance and photoluminescence (PL) spectra. The Judd–Ofelt (J–O) intensity parameters  $\Omega_2 = 0.03 \times 10^{-20} \text{ cm}^2$ ,  $\Omega_4 = 0.22 \times 10^{-20} \text{ cm}^2$ , and  $\Omega_6 = 0.23 \times 10^{-20} \text{ cm}^2$  obtained for the  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphors were used to obtain the total transition probabilities ( $A_T$ ), radiative lifetimes ( $\tau_{\text{rad}}$ ) and branching ratios ( $\beta$ ) for the certain transitions of  $\text{Ho}^{3+}$  ions. Under 310 nm UV excitation, the visible emission spectra were measured, and an intense emission was observed around 541 nm (green region) for all the samples. The emission cross-section  $\sigma_P(\lambda)$  was  $3.22 \times 10^{-21} \text{ cm}^2$  and the branching ratio ( $\beta$ ) was 0.816; these were investigated to capture the optimal concentration of the  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor. The estimated color coordinates were observed in the green region of CIE diagram. Ultimately, the superior properties ( $\sigma_P(\lambda)$ ,  $\beta$ , and color purity) of  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor may make it suitable for green emitting devices.

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## 1. Introduction

Recently, the zircon structured orthovanadate ( $\text{AVO}_4$ , A = trivalent metal ion) has attracted substantial research attention because of its applicability in a wide variety of areas.<sup>1</sup> This material is not only used as a luminescent host material but has also been used in diverse applications in the fields of gas sensing, high power lasers, fuel cell anodes, and counter electrodes in electrochromic devices.<sup>2–4</sup> Yttrium vanadate ( $\text{YVO}_4$ ) is a well-recognized host lattice for rare-earth (RE) activators from the family of  $\text{AVO}_4$ .  $\text{YVO}_4$  shows excellent thermal, mechanical, and optical properties, and it exhibits wide absorption in the UV region; therefore, RE doped phosphors are widely used for optoelectronic application alongwith biosensors and laser devices.<sup>5–12</sup>  $\text{YVO}_4$  shows high birefringence and can be used as a polarizer in the infrared (IR) region.<sup>13</sup>  $\text{YVO}_4$  is also a good up-conversion host due to its low phonon energy ( $\sim 880 \text{ cm}^{-1}$ ). Many studies have examined up-conversion luminescence in  $\text{RE}^{3+}$ -doped and co-doped  $\text{YVO}_4$  phosphors.<sup>14–17</sup>  $\text{YVO}_4:\text{Nd}^{3+}$

performs better than  $\text{YAG}:\text{Nd}^{3+}$ , and it is therefore a good laser excitable host crystal for  $\text{Nd}^{3+}$  ions, and has shown remarkable applicability in laser-diode pumped micro lasers.<sup>18–21</sup>

At present, to conquer multiple color tunability, the phosphor materials are used in energy conversion and materials technology. Due to its efficient energy conversion, the high color purity and thermal stability of  $\text{Eu}^{3+}$ -doped  $\text{YVO}_4$  make it a promising red phosphor for use in a wide range of applications in luminescence and display devices, such as fluorescent lamps, cathode ray tubes, plasma display panels, *etc.*<sup>22–30</sup> Foka *et al.*<sup>31</sup> synthesized  $\text{Dy}^{3+}$ -doped  $\text{YVO}_4$  phosphor via the combustion route and reported white emission with excitation in the UV region. Liu *et al.*<sup>32</sup> synthesized  $\text{YVO}_4:\text{Ln}^{3+}$  (Ln = Eu, Sm, Dy) via the microwave heating method and reported its morphology and photoluminescence (PL) results. Xu *et al.*<sup>33</sup> studied the effects of organic additives on the shape and morphology of  $\text{RE}^{3+}$ -doped  $\text{YVO}_4$  phosphors synthesized via a hydrothermal method. They have found remarkable results, which can be used for tunability in the display panels. Their PL study indicates an efficient multicolor emission that can be useful in lasers and displays related to the optoelectronic devices. Huang *et al.*<sup>34</sup> synthesized ( $\text{Bi}^{3+}$ ,  $\text{RE}^{3+}$ ) co-doped  $\text{YVO}_4$  phosphor by a solid-state reaction technique and reported that the phosphors could improve the efficiency of the power conversion of dye-sensitized and crystalline silicon (c-Si) solar cells. The energy transfer between single and triply doped  $\text{RE}^{3+}$  ions in the  $\text{YVO}_4$  host and its luminescence property have

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recently been studied by Medvedev *et al.*<sup>35</sup> Moreover, all the rare earth ions and their doped systems of the phosphors with  $\text{Ho}^{3+}$  ion serve as good activators due to energy transfer and reflecting the multiple color emission. Zhou *et al.*<sup>36</sup> studied the PL property of red emitting  $\text{YVO}_4:\text{Pr}^{3+}$  phosphor, which is considered to be a good candidate for optical thermal sensing applications.  $\text{YVO}_4$  phosphors have been synthesized through various synthesis methods, such as co-precipitation,<sup>37</sup> solution combustion,<sup>38</sup> solid state reaction,<sup>34</sup> microwave heating,<sup>32</sup> sol-gel,<sup>39,40</sup> microemulsions,<sup>41</sup> solvothermal,<sup>42</sup> ultrasonic,<sup>43</sup> and spray pyrolysis methods.<sup>44</sup> The  $\text{YVO}_4:(\text{Ho}^{3+}, \text{Yb}^{3+})$  phosphors synthesized by different synthesis method and their effect on luminescence have recently been studied.<sup>39</sup> The study indicates that the  $\text{YVO}_4:(\text{Ho}^{3+}, \text{Yb}^{3+})$  phosphor prepared by sol-gel method exhibit better optical properties than the phosphor synthesized by the combustion and solid state reaction methods.

In the present work, most notably, we have successfully synthesized trivalent holmium ( $\text{Ho}^{3+}$ )-doped  $\text{YVO}_4$  phosphors with different doping concentrations. The samples were characterized in terms of their physical and radiative properties by using X-ray diffraction (XRD), scanning electron microscopy (SEM), Fourier transform infrared (FTIR) spectroscopy, ultraviolet-visible (UV-vis) spectrometry, and photoluminescence (PL) techniques.

## 2. Materials preparation and analysis

Phosphor materials of  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  ( $x = 0.01, 0.03, 0.05, 0.07, 0.09, 0.11$ ) were prepared through the sol-gel method. The details of the sample composition and the starting materials along with required weights are given in Table 1. The appropriate amounts of  $\text{Y}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ ,  $\text{NH}_4\text{VO}_3$ ,  $\text{Ho}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ , and citric acid (chelating agent for the metal ions) were weighed and added into a 150 ml glass beaker with 10 ml deionized water. The prepared solution was stirred for 1 hour to ensure uniform homogeneity. Next, the solution was placed in a hot oven until it was completely dried. The dried gels were heated in air at 400 °C for upto 2 hours. The preheated powder samples were ground and again heated in air at 800 °C for 2 hours to acquire the final samples. Fig. 1 shows a flow chart of the synthesis method.

The phase of prepared powders was identified using a RIGAKU (Miniflex-II) XRD spectrometer with  $\text{CuK}\alpha$  light radiation as X-ray. FTIR spectrum in the 4000–400  $\text{cm}^{-1}$  range was obtained on a Thermo Fisher Nicolet (6700 FT-IR)

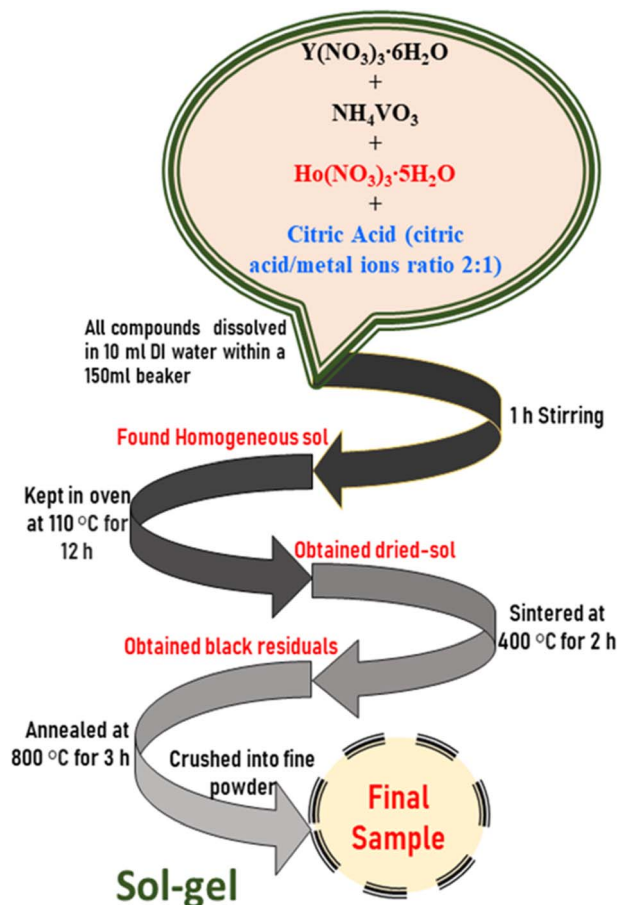


Fig. 1 Flow chart of synthesis method of  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphors.

spectrometer utilizing KBr pellets. The sample surface morphology was imaged using a S-3400, SEM (Hitachi, Japan). The diffuse reflection spectrum was captured with a UV-VIS-NIR (Cary-5000) spectrophotometer. PL characterizations were carried out at room temperature using a RF-5301PC, Shimadzu, fluorescence spectrophotometer with a Xenon flash lamp as an excitation source.

## 3. Results and discussion

Fig. 2 shows the crystal structure of the  $\text{YVO}_4$  compound with different orientations.  $\text{YVO}_4$  is (zircon type structure) crystallized in the tetragonal  $I4_1/a$  space group. In this structure,

Table 1 Detailed information of sample and weight of the required starting materials

Samples	Base materials			
$\text{Y}_{0.99}\text{VO}_4:0.01\text{Ho}^{3+}$	$\text{Y} = 0.7583\text{g}$	$\text{V} = 0.2338\text{g}$	$\text{C.A} = 1.5368\text{g}$	$\text{Ho} = 0.0176\text{g}$
$\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$	$\text{Y} = 0.7430\text{g}$	$\text{V} = 0.2338\text{g}$	$\text{C.A} = 1.5368\text{g}$	$\text{Ho} = 0.0592\text{g}$
$\text{Y}_{0.95}\text{VO}_4:0.05\text{Ho}^{3+}$	$\text{Y} = 0.7277\text{g}$	$\text{V} = 0.2338\text{g}$	$\text{C.A} = 1.5368\text{g}$	$\text{Ho} = 0.0882\text{g}$
$\text{Y}_{0.93}\text{VO}_4:0.07\text{Ho}^{3+}$	$\text{Y} = 0.7123\text{g}$	$\text{V} = 0.2338\text{g}$	$\text{C.A} = 1.5368\text{g}$	$\text{Ho} = 0.1243\text{g}$
$\text{Y}_{0.91}\text{VO}_4:0.09\text{Ho}^{3+}$	$\text{Y} = 0.6970\text{g}$	$\text{V} = 0.2338\text{g}$	$\text{C.A} = 1.5368\text{g}$	$\text{Ho} = 0.1578\text{g}$
$\text{Y}_{0.89}\text{VO}_4:0.11\text{Ho}^{3+}$	$\text{Y} = 0.6817\text{g}$	$\text{V} = 0.2338\text{g}$	$\text{C.A} = 1.5368\text{g}$	$\text{Ho} = 0.1940\text{g}$
$\text{Y} = \text{Y}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$ , $\text{V} = \text{NH}_4\text{VO}_3$ , $\text{C.A} = \text{citric acid}$ , $\text{Ho} = \text{Ho}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$				

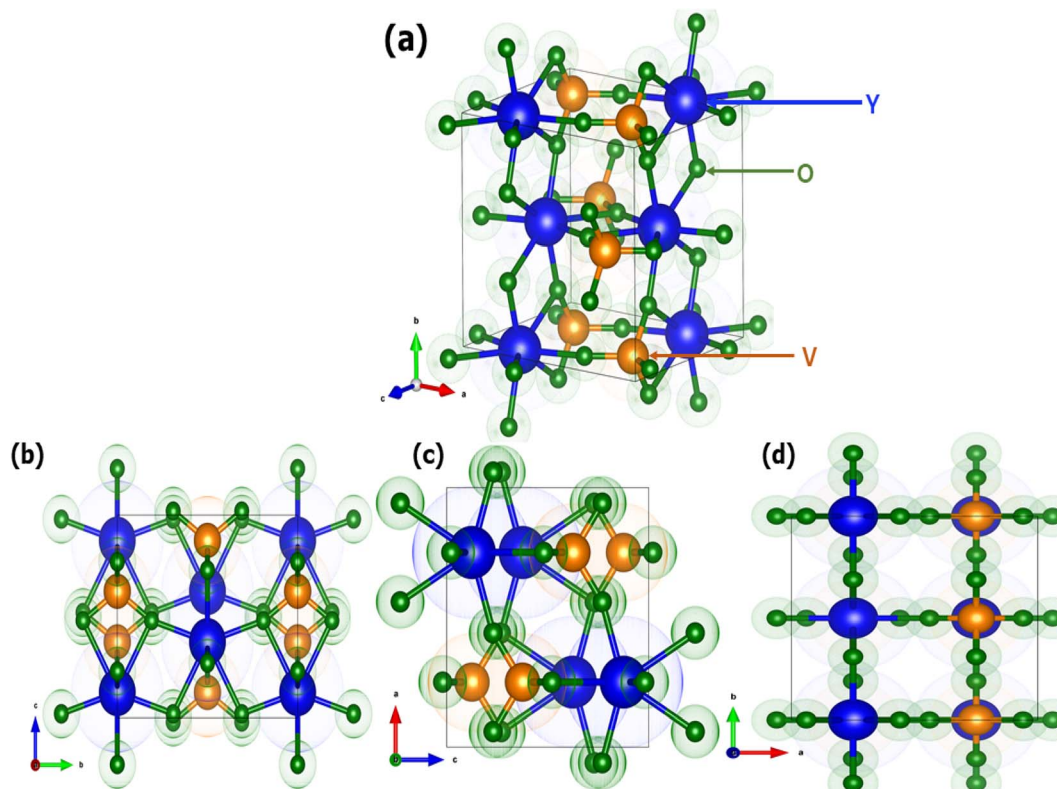


Fig. 2 Crystal structure of  $\text{YVO}_4$  (a) standard orientation and (b–d) three different orientations 100, 010, and 001.

$\text{Y}^{3+}$  atom has  $D_{2d}$  site symmetry and is bonded in 8-coordinate geometry to eight equivalent  $\text{O}^{2-}$  atoms to form an anti-prism. Out of these eight Y–O bonds, four have shorter bond lengths (2.33 Å) whereas the other four have longer bond lengths (2.45 Å). The  $\text{V}^{5+}$  atom is bonded with four equivalent  $\text{O}^{2-}$  atoms to form  $\text{VO}_4^{3-}$  tetrahedra which share edges with  $\text{YO}_8$  dodecahedrons, as shown in Fig. 2. All V–O bond lengths are 1.74 Å. The

$\text{O}^{2-}$  atom is bonded with two equivalent  $\text{Y}^{3+}$  atoms and one  $\text{V}^{5+}$  atom.<sup>45,46</sup> Fig. 3 shows the observed XRD patterns of the  $\text{YVO}_4$  phosphors produced through the sol-gel technique. The observed XRD patterns are counterpart with the standard card of JCPDF 70-1281. No miscellaneous or additional peak is observed in all the XRD patterns, thus suggesting a single-phase material and indicates that the crystal structure of host matrix does not change with the addition of  $\text{Ho}^{3+}$  ion. The ionic radius of  $\text{Ho}^{3+}$  ( $r = 0.89$  Å) was found to be well matched with ionic radius of  $\text{Y}^{3+}$  ( $r = 0.9$  Å, CN = 8), thus indicates that the  $\text{Ho}^{3+}$  is successfully incorporated at the  $\text{Y}^{3+}$  sites. The XRD pattern shows that the prepared  $\text{YVO}_4$  crystallizes in the tetragonal  $I4_1/a$  and space group along with cell parameters  $a = 7.120$  Å,  $c = 6.289$  Å, and volume = 318.82 Å<sup>3</sup>. The average crystallite size can be determined using the formula,

$$D = 0.9\lambda/\beta \cos \theta \quad (1)$$

here,  $\lambda$  (1.540598 Å) is the wavelength of X-ray radiation, and the diffraction angle of Bragg's ( $\theta$ ) and Full width at half maxima (FWHM) is denoted by  $\beta$ . The average crystallite size of the phosphor samples was within the 20–30 nm range.

Fig. 4 presents SEM images of the synthesized  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor. The inhomogeneous morphology and large grain size are observed in the SEM images. SEM images reveal that the phosphor particles morphologies are not uniform, they also show that the phosphor particles are irregular and agglomerated Fig. 4(b) shows an enlarged view of the

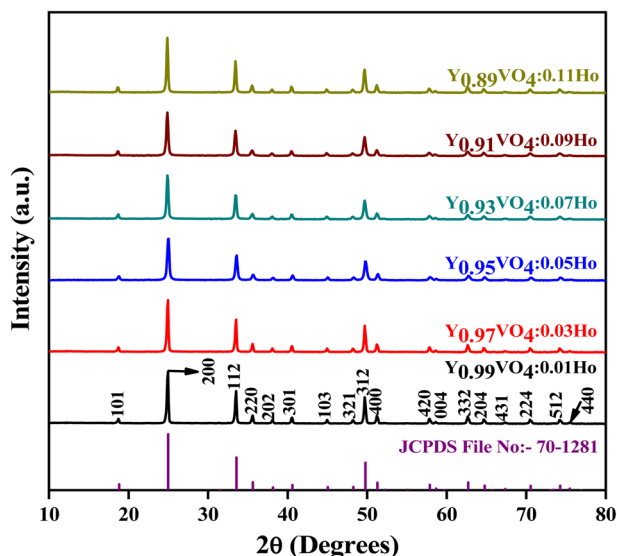


Fig. 3 Powder XRD patterns of  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphors.





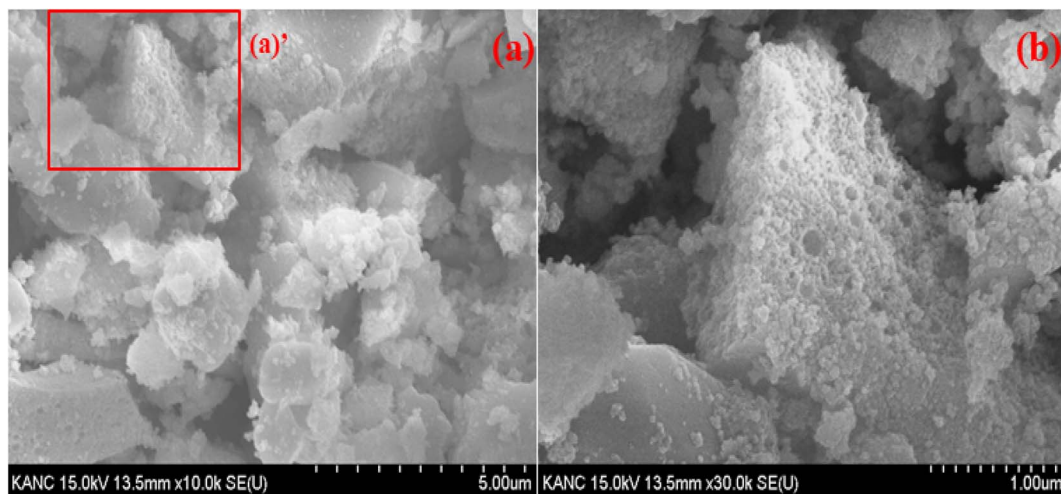


Fig. 4 (a) SEM image of  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor and (b) enlarged view of zone (a)'.

“zone (a)” of Fig. 4(a), wherein the sample consists of voids, and pores that are most likely due to the evolution of large amount of gases during the process of sintering. The grain size distribution is broad, and the average particles size ranges from 100 nm to 500 nm for the  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor. These characteristics suggest that the optical properties are suitable for the industrial use.

Fig. 5 reveals the FTIR spectrum of  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor, which is used to analyze the chemical structure and presence of the functional groups associated with the crystal. The intense band situated at  $814\text{ cm}^{-1}$  is remarkably matched with the characteristic vibrational modes of the V–O group of the tetrahedral structure of  $\text{VO}_4^{3-}$  group.<sup>47</sup> The weak band occurring at  $450\text{ cm}^{-1}$  is attributable to the Y–O group of structural vibrations. The very weak bands observed between  $1100$  to  $1400\text{ cm}^{-1}$  is due to the presence of C–O and nitrate groups alongwith the band between  $1400$  to  $1500\text{ cm}^{-1}$

associated with the vibrations of carboxylate anions of the citric acid. A very weak band observed at  $2350\text{ cm}^{-1}$  corresponds to the asymmetrical stretching modes of  $\text{CO}_2$ .<sup>17</sup> Some very weak bands beyond  $3500\text{ cm}^{-1}$  are also assigned to the water molecules present on the sample surface.

Fig. 6(a) depicts the diffuse reflectance spectrum of  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor. The sharp peaks observed around 421, 458, 469, 478, 489, 544, and 651 nm are associated with the transitions from the ground state of  $^5\text{I}_8$  to the  $^5\text{G}_5$ ,  $^5\text{G}_6 + ^5\text{F}_1$ ,  $^3\text{K}_8$ ,  $^5\text{F}_2$ ,  $^5\text{F}_3$ ,  $^5\text{F}_4 + ^5\text{S}_2$ , and  $^5\text{F}_5$  excited states of  $\text{Ho}^{3+}$  ions, respectively.<sup>48,49</sup> The extracted absorption coefficient ( $\alpha\text{ cm}^{-1}$ ) with the Kubelka–Munk function can be written as:<sup>48</sup>

$$F(R) = \frac{(1 - R)^2}{2R} = \alpha \quad (2)$$

where  $R$  is the sample reflectance. The extracted optical coefficient ( $\alpha\text{ cm}^{-1}$ ) can be calculated by using relation (2). The optical band gap ( $E_g$ ) for the  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor can be estimated using the Tauc relation:<sup>48</sup>

$$F(R) h\nu \approx A(h\nu - E_g)^n \quad (3)$$

where  $A$  is a constant,  $F(R)$  is the absorption coefficient,  $h\nu$  is the energy of the photon, and  $n = 1/2$  (and/or 2) for allowed direct (and/or indirect) transition. Fig. 6(b and c) represents the variation between the  $(F(R) \cdot h\nu)^n$  and  $h\nu$  (eV). By generalizing the linear region of the plot to  $(F(R) \cdot h\nu)^2 = 0$  and  $(F(R) \cdot h\nu)^{1/2} = 0$ , the energy bandgaps for the indirect and direct allowed transitions are found to be  $3.30 \pm 0.24\text{ eV}$  and  $3.59 \pm 0.38\text{ eV}$ , respectively for the  $\text{Ho}^{3+}$  doped  $\text{YVO}_4$  phosphor.

The Judd–Ofelt theory<sup>49,50</sup> is used to characterize the spectroscopic properties of rare earth doped host matrix, for which details are available elsewhere.<sup>51</sup> Therefore, these remarkable outcomes explain the existence theory of the material properties. The doped  $\text{YVO}_4$  phosphor is an excellent host matrix for rare earth doping to support the energy conversion to the  $\text{Ho}^{3+}$  ions. According to Judd–Ofelt theory, the oscillator strengths

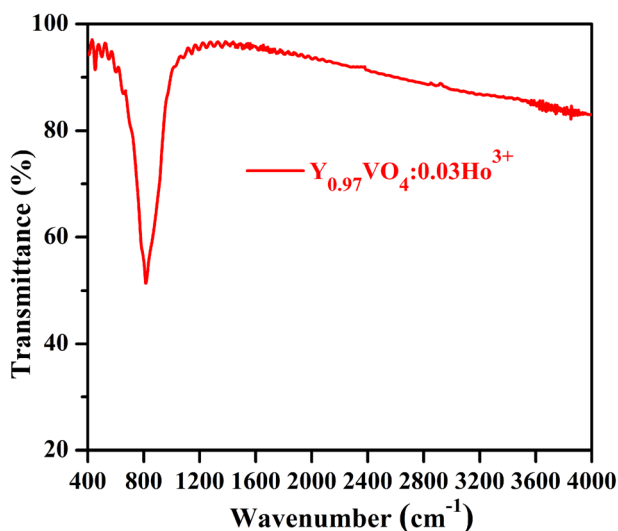


Fig. 5 FT-IR spectrum of  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor.



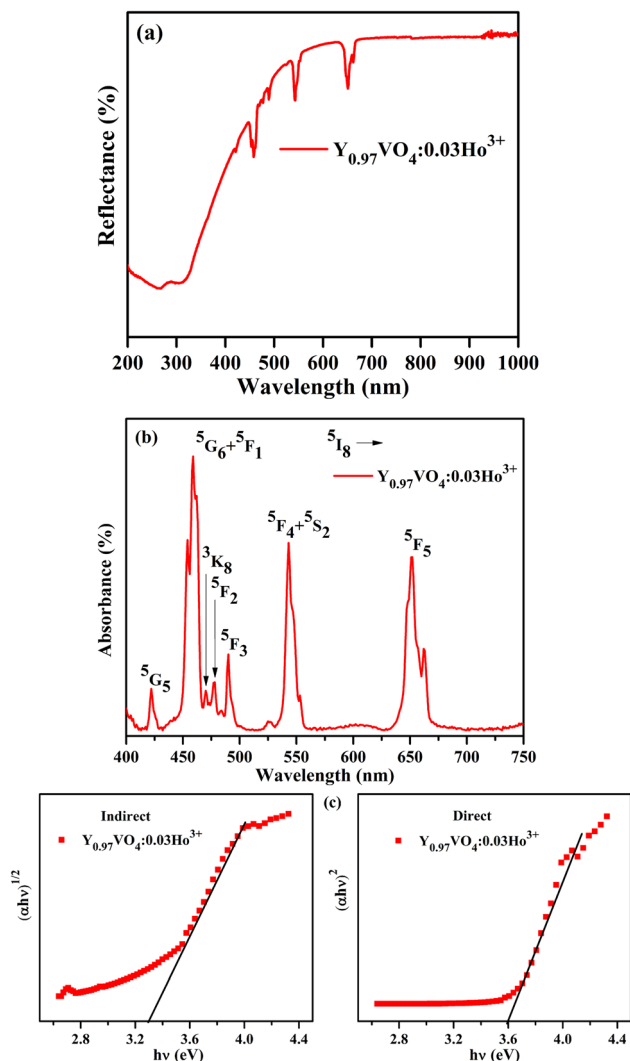


Fig. 6 (a) Diffuse reflection spectrum, (b) absorption spectrum and (c) plots of indirect and direct bandgaps of  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor.

( $f_{\text{cal}}$ ) of an electric dipole multiplets ( $J \rightarrow J'$ ) of  $\text{Ho}^{3+}$  can be expressed as,

$$f_{\text{cal}} = \frac{8\pi^2 m c \cdot (n^2 + 2)^2}{3h(2J + 1)\lambda \cdot 9n} \times \sum_{\lambda=2,4,6} \Omega_{\lambda} |\langle (SL)J \| U^{(\lambda)} \| (S'L')J' \rangle|^2 \quad (4)$$

where  $\Omega_{\lambda}$  ( $\lambda = 2, 4 \text{ \& } 6$ ) are the J–O intensity parameters and  $U^{(\lambda)}$  are the doubly diminished matrix elements taken from ref. 52, and these tensor operators are independent of the host. The measured oscillator strengths ( $f_{\text{meas}}$ ) for the observed absorption bands are calculated using,<sup>53</sup>

$$f_{\text{meas}} = \frac{mc^2}{\pi e^2 \lambda^2 N} \times \frac{2.303}{d} \int \text{OD}(\lambda) d\lambda \quad (5)$$

where  $N$  is the ion concentration (ions/cm<sup>3</sup>),  $d$  is the sample thickness, and  $\text{OD}(\lambda)$  is the optical density as a function of wavelength. Table 2 reports the oscillator strength ( $f_{\text{meas}}$  and  $f_{\text{cal}}$ ) for  $\text{Ho}^{3+}:\text{YVO}_4$  phosphor along with the other hosts.  $\Omega_{\lambda}$

parameters are evaluated through a standard least square fitting method with the known oscillator strengths of the absorption transitions. The magnitude of root mean square ( $\delta_{\text{rms}}$ ) provides information on the quality of the fit, and its formula is,

$$\delta_{\text{rms}} = [(f_{\text{meas}} - f_{\text{cal}})^2 / (p - q)]^{1/2} \quad (6)$$

where  $p$  and  $q$  are the number of transitions employed to the fit and the required fixed parameters, respectively. The observed small r.m.s. deviation (see Table 2) indicates the authenticity of J–O theory.

Table 3 shows the J–O intensity parameters ( $\Omega_2$ ,  $\Omega_4$ , and  $\Omega_6$ ) of various host matrices. It is well known that the  $\Omega_2$  parameter is associated with the covalency of the Ho–O bond, and that the  $\Omega_4$  and  $\Omega_6$  intensity parameters are related to the bulk properties of the host matrix. From Table 3, it can be seen that the magnitude of  $\Omega_2$  in  $\text{YVO}_4$  phosphor is comparable with that in the YAG ( $\text{Y}_3\text{Al}_5\text{O}_{10}$ ) and lower than those in the other reported host matrices, thus indicates lower covalency around the Ho–O bond. Using the J–O intensity parameter,  $\Omega_{\lambda}$  values; the transition probabilities for the radiative transitions ( $A_{\text{rad}}$ ), excited states of radiative lifetimes ( $\tau_{\text{R}}$ ), and branching ratios ( $\beta$ ) are calculated for the  $\text{Ho}^{3+}:\text{YVO}_4$  phosphor using the formula,<sup>53</sup>

$$A_{\text{rad}} = \frac{64\pi^4 \nu^3}{3hc^3(2J + 1)} \frac{n(n^2 + 1)^2}{9} \sum_{\lambda=2,4,6} \Omega_{\lambda} |\langle (SL)J \| U^{(\lambda)} \| (S'L')J' \rangle|^2 \quad (7)$$

$$\tau_{\text{R}} = 1/A_{\text{T}} \quad (8)$$

$$\beta = A_{\text{rad}}/A_{\text{T}} \quad (9)$$

where  $A_{\text{T}}$  is the total radiative transition probability and is estimated by the sum of the  $A_{\text{rad}}$ . Table 4 represents total radiative transition probabilities ( $A_{\text{T}}$ ), radiative lifetime ( $\tau_{\text{rad}}$ ) of certain excited states, and branching ratios for certain emission transitions of  $\text{Ho}^{3+}$  ions in  $\text{YVO}_4$  phosphor. From Table 4, it can be seen that the magnitudes of radiative lifetime for the excited levels are in the order of  ${}^3\text{K}_8 > {}^5\text{F}_5 > {}^5\text{F}_2 > {}^5\text{F}_3 > {}^5\text{F}_4 > {}^3\text{H}_5 > {}^5\text{G}_6$ . The observed magnitudes of the radiative lifetimes of  $\text{Ho}^{3+}:\text{YVO}_4$  are comparatively higher than those of the reported  $\text{Ho}^{3+}$  doped hosts:  $\text{CaScO}_4$ ,<sup>54</sup>  $\text{SrLaGaO}_4$ ,<sup>55</sup>  $\text{Y}_2\text{O}_3$ ,<sup>58</sup> and  $\text{Lu}_2\text{SiO}_5$ ,<sup>60</sup> respectively.

Fig. 7(a) represents the PL excitation spectra of  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphor materials in the wavelength range from 220 to 530 nm that are obtained by observing the emission wavelength at 541 nm. The broad excitation band appearing between the ~220 and 350 nm wavelength range is the charge transfer (CT) transition (at 310 nm) from the oxygen ligand ( $\text{O}^{2-}$ ) to the central vanadium atom ( $\text{V}^{5+}$ ) within the  $\text{VO}_4^{3-}$  group, as well as the CT band (at 270 nm) from  $\text{O}^{2-}$  to  $\text{Ho}^{3+}$  ions.<sup>32,44,61,62</sup> Some sharp excitation peaks appear at ~360, 420, 450, 456, 468, 475, and 487 nm, and these are attributed to the transitions from the ground state  ${}^5\text{I}_8$  to  ${}^3\text{H}_6$ ,  ${}^5\text{G}_5$ ,  ${}^5\text{F}_1$ ,  ${}^5\text{G}_6$ ,  ${}^3\text{K}_8$ ,  ${}^5\text{F}_2$ , and  ${}^5\text{F}_3$  of the excited states of  $\text{Ho}^{3+}$  ions, respectively.<sup>63</sup> Among all these observed sharp peaks, the 456 nm peak is the most prominent.



**Table 2** Measured and calculated oscillator strengths ( $f_{\text{exp}}$  and  $f_{\text{cal}}$ ) for  $\text{Ho}^{3+}$  ions in the  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor

[ $S'LJ'$ ] manifold	$\lambda$ (nm)	$\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$ <sup>a</sup>		CaSc <sub>2</sub> O <sub>4</sub> (ref. 54)	SrLaGaO <sub>4</sub> (ref. 55)	Y <sub>3</sub> Al <sub>5</sub> O <sub>10</sub> (ref. 56)
		$f_{\text{meas}} \times 10^{-6}$	$f_{\text{cal}} \times 10^{-6}$	$f_{\text{meas}} \times 10^{-6}$	$f_{\text{meas}} \times 10^{-6}$	$f_{\text{meas}} \times 10^{-6}$
<sup>5</sup> G <sub>5</sub>	421	0.05	0.35	1.11	—	3.61
<sup>5</sup> G <sub>6</sub> + <sup>5</sup> F <sub>1</sub>	458	0.73	0.74	5.47	20.7	5.83
<sup>3</sup> K <sub>8</sub>	469	0.01	0.11	—	0.29	4.03
<sup>5</sup> F <sub>2</sub>	478	0.03	0.11	—	0.10	—
<sup>5</sup> F <sub>3</sub>	489	0.09	0.20	—	0.92	—
<sup>5</sup> S <sub>2</sub> + <sup>5</sup> F <sub>4</sub>	544	0.48	0.62	1.73	3.86	4.72
<sup>5</sup> F <sub>5</sub>	651	0.68	0.44	2.12	3.37	3.86
$\delta_{\text{rms}}$	0.45	—	—	0.14	5.36	0.89

<sup>a</sup> Present work.**Table 3** Judd–Ofelt intensity parameters ( $\Omega_\lambda$ ,  $\lambda = 2, 4$  &  $6$ ) for  $\text{Ho}^{3+}$  ions in the host matrices

Host matrix	$\Omega_2$	$\Omega_4$	$\Omega_6$
$\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$ <sup>a</sup>	0.03	0.22	0.23
CaSc <sub>2</sub> O <sub>4</sub> (ref. 54)	3.78	5.17	1.92
SrLaGaO <sub>4</sub> (ref. 55)	1.25	0.42	1.80
Y <sub>3</sub> Al <sub>5</sub> O <sub>10</sub> (ref. 56)	0.04	2.67	1.89
SrLaGa <sub>3</sub> O <sub>7</sub> (ref. 57)	2.23	0.85	1.81
Y <sub>2</sub> O <sub>3</sub> (ref. 58)	0.45	0.59	0.32
LiYF <sub>4</sub> (ref. 59)	1.16	2.24	2.09

<sup>a</sup> Present work.**Table 4** Radiative properties of certain excited levels/transitions of  $\text{Ho}^{3+}$  ions in the  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor

[ $S'LJ'$ ] manifold	$A_{\text{T}}$ (s <sup>-1</sup> )	$\tau_{\text{rad}}$ (μs)
<sup>3</sup> H <sub>5</sub>	1497	668
<sup>5</sup> G <sub>6</sub>	1592	628
<sup>3</sup> K <sub>8</sub>	159	6297
<sup>5</sup> F <sub>2</sub>	867	1153
<sup>5</sup> F <sub>3</sub>	1060	943
<sup>5</sup> F <sub>4</sub>	1082	924
<sup>5</sup> F <sub>5</sub>	542	1845
Transition	$\beta$	
<sup>3</sup> H <sub>5</sub> → <sup>5</sup> I <sub>8</sub>	0.423	
<sup>5</sup> G <sub>6</sub> → <sup>5</sup> I <sub>8</sub>	0.752	
<sup>3</sup> K <sub>8</sub> → <sup>5</sup> I <sub>8</sub>	0.880	
<sup>5</sup> F <sub>2</sub> → <sup>5</sup> I <sub>8</sub>	0.566	
<sup>5</sup> F <sub>3</sub> → <sup>5</sup> I <sub>8</sub>	0.524	
<sup>5</sup> F <sub>4</sub> → <sup>5</sup> I <sub>8</sub>	0.816	
<sup>5</sup> F <sub>5</sub> → <sup>5</sup> I <sub>8</sub>	0.775	

Under UV excitation of 310 nm, the PL emission spectra of  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphors are measured from the 450 nm to 700 nm wavelength region, and the results are represented in Fig. 7(b). The emission spectra show intense green emissions around 541, 546, and 551 nm, which are attributed to the (<sup>5</sup>S<sub>2</sub> + <sup>5</sup>F<sub>4</sub>) → <sup>5</sup>I<sub>8</sub> transitions of  $\text{Ho}^{3+}$  ion. A few small emission peaks are

observed around 475 nm (<sup>5</sup>F<sub>2</sub> → <sup>5</sup>I<sub>8</sub>), 485 nm (<sup>5</sup>F<sub>3</sub> → <sup>5</sup>I<sub>8</sub>), 526 nm (<sup>5</sup>G<sub>5</sub> → <sup>5</sup>I<sub>7</sub>), and 650 and 661 nm (<sup>5</sup>F<sub>5</sub> → <sup>5</sup>I<sub>8</sub>) wavelengths.<sup>63,64</sup> Fig. 8 shows the energy level diagram of  $\text{Ho}^{3+}$  ion with possible transitions. Under UV excitation at 310 nm, the excitation energy is first absorbed by  $\text{VO}_4^{3-}$  group, which excites the electrons from the filled oxygen 2p levels in the valence band to the empty vanadium 3d levels of the conduction band. The emission from the  $\text{VO}_4^{3-}$  group is not observed in this phosphor. Instead of emission from the  $\text{VO}_4^{3-}$  group, the resonant energy transfer takes place from the  $\text{VO}_4^{3-}$  group to the <sup>5</sup>G<sub>4</sub> level of the  $\text{Ho}^{3+}$  ion. From <sup>5</sup>G<sub>4</sub> level, the non-radiative transitions take place to lower lying levels of  $\text{Ho}^{3+}$  ion and due to this, various emissions take place at different wavelengths as indicated in Fig. 8.

In general, the doping concentration could affect the efficiency or performance of the phosphor, so it is necessary to analyze the effect of  $\text{Ho}^{3+}$  ion concentration on the PL intensity. In the present analysis, the concentration of  $\text{Ho}^{3+}$  ion in the  $\text{YVO}_4$  host is varied from 0.01 to 0.11 mol, and the resultant PL intensity as a function of  $\text{Ho}^{3+}$  ion concentration is shown in Fig. 9. The PL intensity is maximum at  $x = 0.03$ , and it then decreases due to the effect of concentration quenching. This result suggests that the optimum concentration of  $\text{Ho}^{3+}$  ion in the  $\text{YVO}_4$  host is 0.03 mol. The concentration quenching depends on the critical distance ( $R_c$ ) between the activator ion and the quenching site. According to Blasse,  $R_c$  can be calculated using the following expression:<sup>65</sup>

$$R_c = 2 \left( \frac{3V}{4\pi X_c N} \right)^{1/3} \quad (10)$$

here,  $V$  = volume of unit cell,  $X_c$  = optimal concentration of  $\text{Ho}^{3+}$  ions, and  $N$  = number of dopant sites available in the unit cell. In our case,  $V = 318.82 \text{ \AA}^3$ ,  $X_c = 0.03$ , and  $N = 4$ . From eqn (10), this equates to a critical distance of 17.18 Å, which is greater than 5 Å, thus indicating a smaller chance of energy transfer *via* exchange interaction. Therefore, the process of energy transfer occurs *via* electric multipolar interaction, which causes concentration quenching of  $\text{Ho}^{3+}$  ion in the prepared phosphors.

The peak stimulated emission cross-section ( $\sigma_p$ ) for the (<sup>5</sup>S<sub>2</sub> + <sup>5</sup>F<sub>4</sub>) → <sup>5</sup>I<sub>8</sub> transition of  $\text{Ho}^{3+}$  ion for the optimum



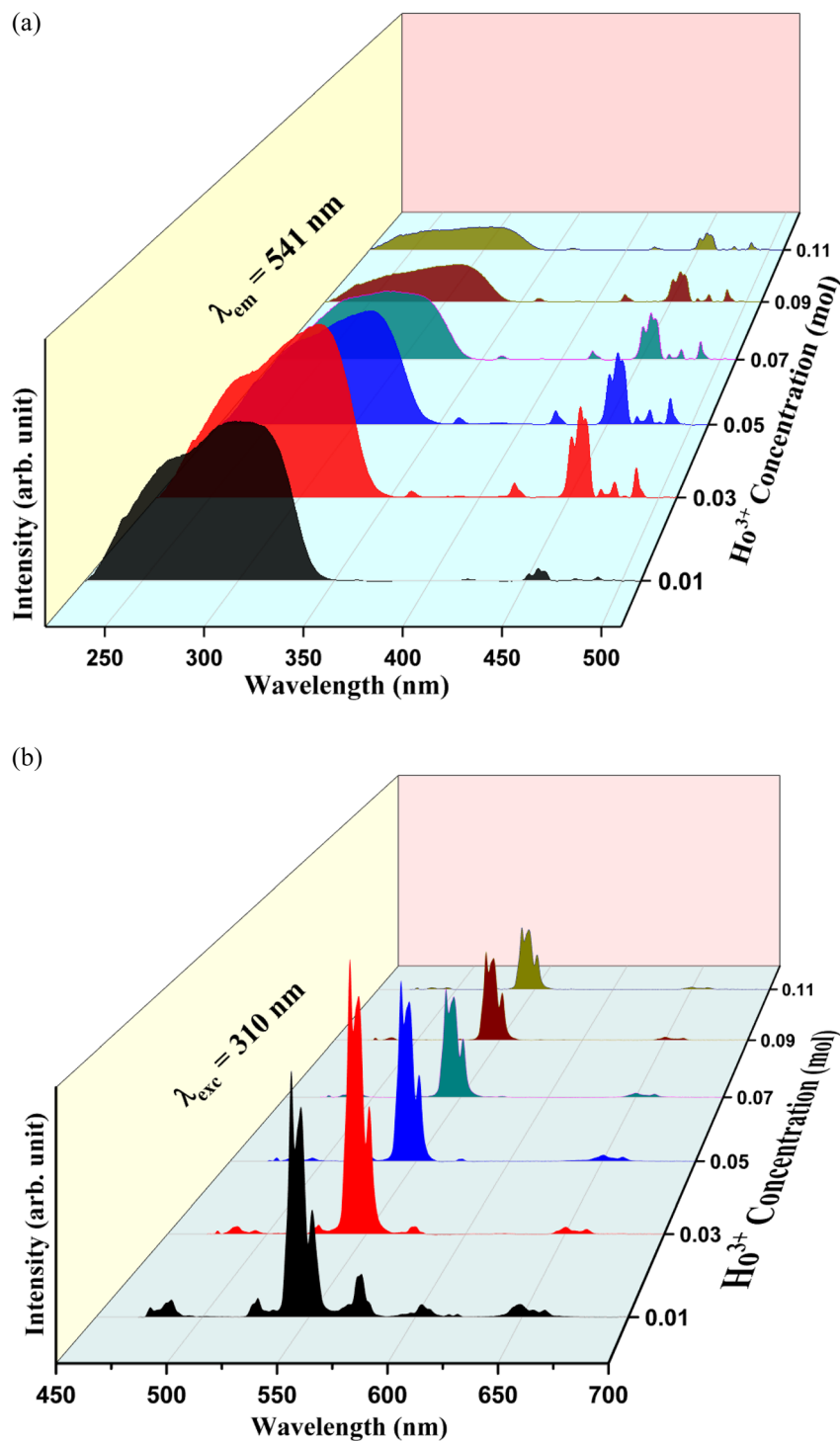


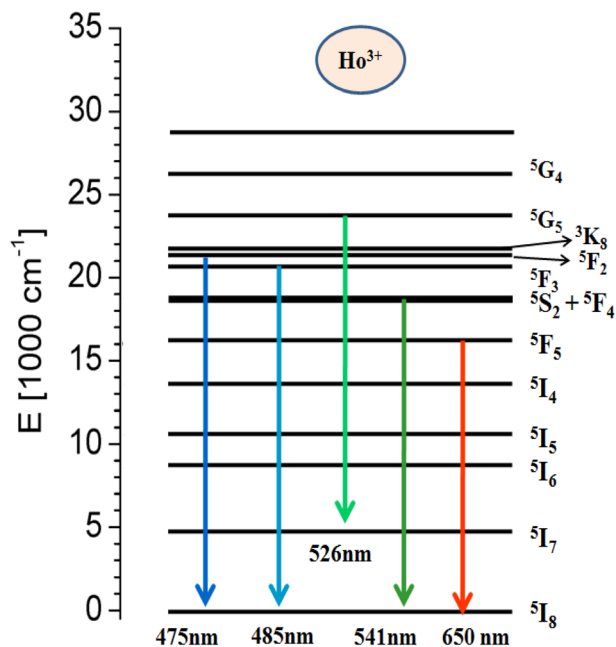
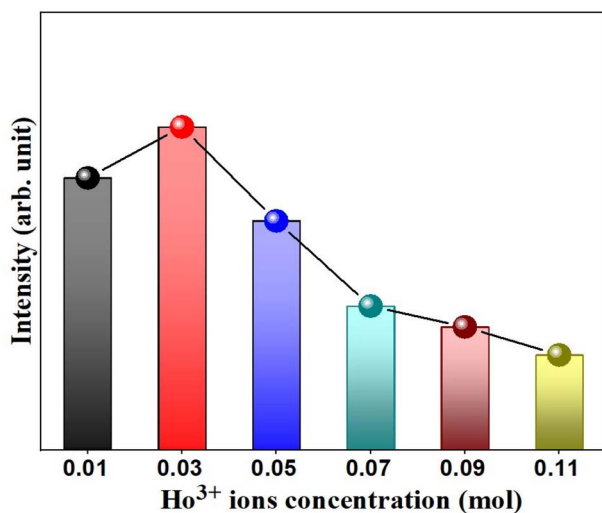
Fig. 7 Photoluminescence spectra of  $Y_{1-x}VO_4:xHo^{3+}$  phosphors: (a) excitation spectra ( $\lambda_{emi} = 541$  nm) and (b) emission spectra ( $\lambda_{exc} = 310$  nm).

( $Y_{0.97}VO_4:0.03Ho^{3+}$ ) phosphor is also obtained from the following relation<sup>53</sup>

$$\sigma_p(\lambda) = \frac{\lambda_p^4}{8\pi cn^2 \Delta\lambda_{eff}} \cdot A_{rad} \quad (11)$$

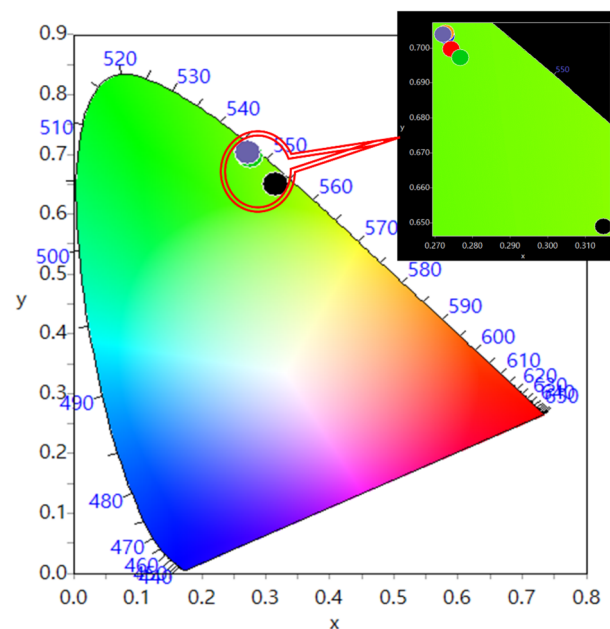
Here,  $\lambda_p$  and  $\Delta\lambda_{eff}$  are the peak wavelength and effective line width of the emission band, respectively. The estimated  $\Delta\lambda_{eff}$  and  $\sigma_p(\lambda)$  are 6.33 nm and  $3.22 \times 10^{-21} \text{ cm}^2$ , respectively for the ( $^5S_2 + ^5F_4$ )  $\rightarrow$   $^5I_8$  transition of  $Ho^{3+}$  ions in  $YVO_4$  phosphor. The values of  $\sigma_p(\lambda)$  ( $3.22 \times 10^{-21} \text{ cm}^2$ ) and  $\beta$  (0.816) for the green



Fig. 8 Schematic energy level diagram of  $\text{Ho}^{3+}$  ion.Fig. 9 Variation of emission intensity ( $I_{541\text{nm}}$ ) as a function of  $\text{Ho}^{3+}$  concentration in the  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphor.

emission of  $\text{Ho}^{3+}:\text{YVO}_4$  are larger than  $\text{Y}_2\text{O}_3:\text{Ho}^{3+}$  ( $\sigma_p(\lambda) = 1.32 \times 10^{-21} \text{ cm}^2$ ;  $\beta = 0.65$ ),<sup>58</sup> which suggests that the  $\text{Ho}^{3+}:\text{YVO}_4$  phosphor is a desirable optical material for low threshold and high gain applications. These remarkable results provide information on energy transfer from  $\text{VO}_4^{3-}$  group to  $\text{Ho}^{3+}$  ions.

The CIE (Commission International de l'Eclairage) chromaticity coordinates of the prepared  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphor materials are represented in Fig. 10. The results show that the CIE coordinates fall in the green region.<sup>66,67</sup> The color coordinates and color purity of the prepared phosphors are summarized in Table 5.

Fig. 10 CIE diagram for  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphors.Table 5 Commission International de l'Eclairage (CIE) color coordinates and color purity for the  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphors using PL emission at  $\lambda_{\text{exc}} = 310 \text{ nm}$ 

$\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$	$\lambda_{\text{exc}} = 310 \text{ nm}$		Color purity (%)
	x	y	
$x = 0.01$	0.315	0.649	84.27(●)
$x = 0.03$	0.273	0.704	98.64(●)
$x = 0.05$	0.274	0.700	97.61(●)
$x = 0.07$	0.276	0.700	97.56(●)
$x = 0.09$	0.277	0.697	96.78(●)
$x = 0.11$	0.282	0.699	97.18(●)

## 4. Conclusions

In summary, we report herein the spectroscopic properties of a homogenous series of  $\text{Y}_{1-x}\text{VO}_4:x\text{Ho}^{3+}$  phosphor materials synthesized using the sol-gel synthesis procedure. SEM analysis shows the irregular particles size of the phosphor. The preparation of a single-phase compound was confirmed by the XRD patterns. The optical transitions of  $\text{Ho}^{3+}$  ions are identified through diffuse reflectance spectrum. Using these absorption bands, the oscillator strength of the bands ( $f_{\text{exp}}$  &  $f_{\text{cal}}$ ), the Judd-Ofelt parameters ( $\Omega_\lambda$ ,  $\lambda = 2, 4, \& 6$ ), total radiative transition probabilities ( $A_T$ ), radiative lifetimes ( $\tau_{\text{rad}}$ ), and branching ratios ( $\beta$ ) for the transition of  $\text{Ho}^{3+}$  ions in  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor are reported. The energy conversion; and optical direct and indirect bandgaps are observed for  $\text{YVO}_4$  phosphor doped with  $\text{Ho}^{3+}$  ions. In this approach, the energy transfer to the  $\text{Ho}^{3+}$  ions supports the results of the spectroscopic analyses. The phosphor exhibited a strong excitation band in the UV region at 310 nm while monitoring emission at 541 nm. Upon 310 nm



excitation, all the synthesized phosphors showed intense PL emission in the green region. Concentration quenching of  $\text{Ho}^{3+}$  ion was observed after 0.03 mol and this was mainly attributed to the electric multipolar interaction process. The  $\text{Y}_{0.97}\text{VO}_4:0.03\text{Ho}^{3+}$  phosphor showed good emissive properties with very high color purity, suggesting that this material would be useful for green emitting devices.

## Conflicts of interest

There are no conflicts to declare.

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## References

- V. Panchal, S. Lopez-Moreno, D. Sanatamaria-Perez, D. Errandonea, F. J. Manjon, P. Rodriguez-Hernandez, A. Munoz, S. N. Achary and A. K. Tyagi, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 02411.
- V. Panchal, D. Errandonea, A. Segura, P. Rodriguez-Hernandez, A. Munoz, S. Lopez-Moreno and M. Bettinelli, *J. Appl. Phys.*, 2011, **110**, 43723.
- N. Deligne, V. Gonze, D. Bayot and M. Devillers, *Eur. J. Inorg. Chem.*, 2008, **6**, 896–902.
- E. V. Tsipis, V. V. Kharton, N. P. Vyshatko, A. L. Shaula and J. R. Frade, *J. Solid State Chem.*, 2003, **176**, 47–56.
- Z. Xu, X. Kang, C. Li, Z. Hou, C. Zhang, D. Yang, G. Li and J. Lin, *Inorg. Chem.*, 2010, **49**, 6706.
- Z. Y. Hou, P. P. Yang, C. X. Li, L. L. Wang, H. Z. Lian, Z. W. Quan and J. Lin, *Chem. Mater.*, 2008, **20**, 6686.
- F. He, P. Yang, N. Niu, W. Wang, S. Gai, D. Wang and J. Lin, *J. Colloid Interface Sci.*, 2010, **343**, 71.
- P. Huang, D. Chen and Y. Wang, *J. Alloys Compd.*, 2011, **509**, 3375.
- L. Xie, H. Song, Y. Wang, W. Xu, X. Bai and B. Dong, *J. Phys. Chem. C*, 2010, **114**, 9975.
- L. Li, M. Zhao, W. Tong, X. Guan, G. Li and L. Yang, *Nanotechnology*, 2010, **21**, 195601.
- M. Yu, J. Lin and J. Fang, *Chem. Mater.*, 2005, **17**, 1783–1791.
- A. Bao, H. Lai, Y. Yang, Z. Liu, C. Tao and H. Yang, *J. Nanopart. Res.*, 2010, **12**, 635–643.
- E. A. Maunders and L. G. DeShazer, *J. Opt. Soc. Am.*, 1971, **61**, 684.
- F. S. Ermeneux, R. Moncorge, P. Kabro, J. A. Capobianco, M. Bettinelli and E. Cavalli, *Adv. Solid State Lasers*, OSA, Washington, DC, 1996, p. SM9.
- J. Sun, J. Zhu, X. Liu and H. Du, *Mater. Res. Bull.*, 2013, **48**, 2175–2179.
- J. A. Coppabianco, P. Kabro, F. S. Ermeneux, R. Moncorge, M. Bettinelli and E. Cavalli, *Chem. Phys.*, 1997, **214**, 329–340.
- M. Mahata, S. Tiwari, S. Mukherjee, K. Kumar and V. Rai, *J. Opt. Soc. Am. B*, 2014, **31**, 1814–1821.
- J. E. Bernard and A. J. Alcock, *Opt. Lett.*, 1993, **18**, 968.
- G. C. Bowkett, G. W. Baxter, D. J. Booth, T. Taira, H. Teranishi and T. Kobayashi, *Opt. Lett.*, 1994, **19**, 957.
- D. Shen, A. Liu, J. Song and K. Ueda, *Appl. Opt.*, 1998, **37**, 7785.
- X. Li, H. Xu, R. Yan, Y. Jiang, R. Fan, Z. Dong and D. Chen, *Optik*, 2021, **228**, 165789.
- A. K. Levine and F. C. Palilla, *Appl. Phys. Lett.*, 1964, **5**, 118–120.
- A. Bril, W. L. Wanmaker and J. Broos, *J. Chem. Phys.*, 1965, **43**, 311.
- L. R. Singh and R. S. Ningthoujam, *J. Appl. Phys.*, 2010, **107**, 104304–104306.
- W. M. Yen, S. Shionoya and H. Yamamoto, *Phosphor Handbook*, CRC Press, Boca-Raton, 2<sup>nd</sup> edn, 1999.
- C. Hsu and R. C. Powell, *J. Lumin.*, 1975, **10**, 273–293.
- M. Wu, S. Choi and H. K. Jung, *Mater. Res. Bull.*, 2016, **78**, 20–25.
- U. Rambabu, D. P. Amalnerkar, B. B. Kale and S. Buddhudu, *Mater. Res. Bull.*, 2000, **35**, 929–936.
- L. Shirmane, C. Feldmann and V. Pankratov, *Phys. B*, 2017, **504**, 80–85.
- S. M. Rafiaei and M. Shokouhimehr, *Mater. Chem. Phys.*, 2019, **229**, 431–436.
- K. E. Foka, B. F. Dejene and H. C. Swart, *Phys. B*, 2016, **480**, 95–99.
- Y. Liu, H. Xiong, N. Zhang, Z. Leng, R. Li and S. Gan, *J. Alloys Compd.*, 2015, **653**, 126–134.
- Z. Xu, X. Kang, C. Li, Z. Hou, C. Zhang, D. Yang, G. Li and J. Lin, *Inorg. Chem.*, 2010, **49**, 6706–6715.
- X. Y. Huang, J. X. Wang, D. C. Yu, S. Ye, Q. Y. Zhang and X. W. Sun, *J. Appl. Phys.*, 2011, **109**, 113526.
- V. A. Medvedev, D. V. Mamonova, I. E. Kolesnikov, A. R. Khokhlova, M. D. Mikhailov and A. A. Manshina, *Inorg. Chem. Commun.*, 2020, **118**, 107990.
- H. Zhou, W. Gao, P. Cai, B. Zhang and S. Li, *Solid State Sci.*, 2020, **104**, 106283.
- Y. H. Li and G. Y. Hong, *J. Solid State Chem.*, 2005, **175**, 645–649.
- S. Ekamparam and K. C. Patil, *J. Alloys Compd.*, 1995, **217**, 104–107.
- A. Dwivedi, E. Rai, D. Kumar and S. B. Rai, *ACS Omega*, 2019, **4**, 6903–6913.
- M. Yu, J. Lin, Z. Wang, J. Fu, S. Wang, H. J. Zhang and Y. C. Han, *Chem. Mater.*, 2002, **14**, 2224–2231.
- L. D. Sun, Y. X. Zhang, J. Zhang, C. H. Yan, C. S. Liao and Y. Q. Lu, *Solid State Commun.*, 2002, **124**, 35–38.
- G. Jia, Y. Song, M. Yang, Y. Huang, L. Zhang and H. You, *Opt. Mater.*, 2009, **31**, 1032–1037.
- L. Zhu, J. Y. Li, Q. Li, X. D. Liu, J. Meng and X. Q. Cao, *Nanotechnology*, 2007, **18**, 055604–055609.
- Y. H. Zhou and J. Lin, *Opt. Mater.*, 2005, **27**, 1426–1432.
- M. R. Dolgos, A. M. Paraskos, M. W. Stoltzfus, S. C. Yarnell and P. M. Woodward, *J. Solid State Chem.*, 2009, **182**, 1964–1971.
- W. O. Milligan and L. W. Vernon, *J. Phys. Chem. A*, 1952, **56**, 145–147.



- 47 Y. Sun, H. Liu, X. Wang, X. Kon and H. Zhang, *Chem. Mater.*, 2006, **18**, 2726–2732.
- 48 C. Shivakumara, R. Saraf and P. Halappa, *Dyes Pigm.*, 2016, **126**, 154–164.
- 49 B. R. Judd, *Phys. Rev.*, 1962, **127**, 750.
- 50 G. S. Ofelt, *J. Chem. Phys.*, 1962, **37**, 511.
- 51 A. A. Kaminskii, *Crystalline Lasers*, CRC Press, 1996.
- 52 W. T. Carnall, H. Crosswhite and H. M. Crosswhite, *Energy Level Structure and Transition Probabilities of the Trivalent Lanthanides in Laf*, Argonne National Laboratory, Argonne, IL, 1975.
- 53 M. Seshadri, E. F. Chillece, J. D. Marconi, F. A. Sigoli, Y. C. Ratnakaram and L. C. Barbosa, *J. Non-Cryst. Solids*, 2014, **402**, 141.
- 54 Ş. Georgescu, A. Ştefan, O. Toma and A.-M. Voiculescu, *J. Lumin.*, 2015, **162**, 174–179.
- 55 M. Kaczkan and M. Malinowski, *Materials*, 2021, **14**, 3831.
- 56 M. Malinowski, Z. Frukacz, M. Szuflinska, A. Wnuk and M. Kaczkan, *J. Alloys Compd.*, 2000, **300–301**, 389–394.
- 57 M. Kaczkan, I. Pracka and M. Malinowski, *Opt. Mater.*, 2004, **25**, 345–352.
- 58 V. Singh, V. K. Rai, B. Voss, M. Haase, R. P. S. Chakradhar, D. T. Naidu and S. H. Kim, *Spectrochim. Acta, Part A*, 2013, **109**, 206–212.
- 59 B. M. Walsh, G. W. Grew and N. P. Barnes, *J. Phys.: Condens. Matter*, 2005, **17**, 7643–7665.
- 60 Q. Dong, G. Zhao, J. Chen, Y. Ding, B. Yao and Z. Yu, *Opt. Mater.*, 2010, **32**, 873–877.
- 61 S. R. Bullock, B. R. Reddy, P. Venkateswarlu and S. K. N. Stevenson, *J. Opt. Soc. Am. B*, 1997, **14**, 553–559.
- 62 A. Huignard, V. Buissette, A. Franville, T. Gacoin and J. Boilot, *J. Phys. Chem. B*, 2003, **107**, 6754–6759.
- 63 I. E. Kolesnikov, A. A. Kalinichev, M. A. Kurochkin, D. V. Mamonova, E. Y. Kolesnikov and E. Lahderanta, *J. Phys. Chem. C*, 2019, **123**, 5136–5143.
- 64 V. Singha, G. Lakshminarayana, A. Wagh and N. Singh, *Optik*, 2020, **207**, 164284.
- 65 G. Blasse, *Phys. Lett. A*, 1968, **28**, 444–445.
- 66 Monika, R. S. Yadav, A. Rai and S. B. Rai, *Sci. Rep.*, 2021, **11**, 4148.
- 67 P. Tadge, I. R. Martín, S. B. Rai, S. Sapra, T. M. Chen, V. Lavín, R. S. Yadav and S. Ray, *J. Lumin.*, 2022, **252**, 119261.

