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# CuO nanorod arrays by gas-phase cation exchange for efficient photoelectrochemical water splitting†

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CuO has been considered a promising candidate for photoelectrochemical water splitting electrodes owing to its suitable bandgap, favorable band alignments, and earth-abundant nature. In this paper, a novel gas-phase cation exchange method was developed to synthesize CuO nanorod arrays by using ZnO nanorod arrays as the template. ZnO nanorods were fully converted to CuO nanorods with aspect ratios of 10–20 at the temperature range from 350 to 600 °C. The as-synthesized CuO nanorods exhibit a photocurrent as high as 2.42 mA cm<sup>-2</sup> at 0 V vs. RHE (reversible hydrogen electrode) under 1.5 AM solar irradiation, demonstrating the potential as the photoelectrode for efficient photoelectrochemical water splitting. Our method provides a new approach for the rational fabrication of high-performance CuO-based nanodevices.

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## 1. Introduction

Photoelectrochemical (PEC) water splitting has emerged as a sustainable technology for solar energy conversion to meet the growing demand for clean energy. Despite the significant progress that has been made over the past decade, improvement of the effectiveness and robustness of PEC cells to meet the requirement for widespread application remains a challenge.<sup>1–3</sup> In this regard, developing high-performance electrode materials is highly desirable. CuO, as an important p-type semiconductor, has been demonstrated as a promising candidate for PEC water splitting owing to its suitable bandgap of around 1.2 eV, which guarantees efficient solar light absorption in the visible light region, thus having a high theoretical maximum photocurrent of 35 mA cm<sup>-2</sup> under standard AM 1.5 irradiation.<sup>4,5</sup> In addition, the p-type semiconductor nature of CuO provides advantages over n-type semiconductors in terms of hydrogen generation since electrons can be injected into the electrolyte directly. On the other hand, the electrons on n-type semiconductors need to migrate through an external circuit, thus causing a potential energy loss.<sup>6</sup> The conduction band edge of CuO lies 0.2 eV more negative than the hydrogen evolution potential, which is favorable as photocatalysts for hydrogen generation.<sup>7</sup> Furthermore, the earth-abundance of the

copper element makes the large-scale and low-cost fabrication of CuO photoelectrode possible.<sup>8</sup>

For PEC water splitting, one of the most efficient strategies to improve its performance is to construct nanostructures, especially one-dimensional nanostructured photoelectrodes, such as nanowires, nanorods, and nanotubes.<sup>9–11</sup> The well-aligned one-dimensional nanostructures provide efficient light absorption, short charge carrier diffusion paths, as well as a large surface area resulting in improved exposure of the catalytic sites.<sup>12</sup> For the synthesis of CuO one-dimensional nanostructures, thermal oxidation is the most widely used method due to its simplicity.<sup>13</sup> It has been recently demonstrated that CuO nanowires synthesized on copper foils by the thermal oxidation method present high performance for PEC water splitting.<sup>14,15</sup> However, the growth of CuO nanowires by the thermal oxidation method was usually confined to limited choices of substrates, such as copper or specially fabricated copper-coated substrates, which hindered its broader applications.<sup>16,17</sup> In addition, the inevitable cracking and exfoliation problems make the large-area synthesis difficult.<sup>18</sup> Therefore, it is highly desirable to develop novel methods for the direct synthesis of CuO one-dimensional nanostructures. Recently, by using ZnO nanostructures as templates, the gas-phase cation exchange method has been adopted as an efficient way for the synthesis of various metal oxide nanostructures, such as Mn<sub>3</sub>O<sub>4</sub>, NiO, and CoO.<sup>19–21</sup> The exchange process between the metal cations and Zn cations which is facilitated by the thermal disturbance well preserves the original morphology and the crystallinity structure of the ZnO nanostructures, thus has shown great potential in numerous energy-related applications, including water splitting, supercapacitors, and batteries.<sup>22–24</sup>

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Inspired by the above results, we developed a gas-phase cation exchange method for the synthesis of CuO nanorod arrays and demonstrated its potential for efficient PEC water splitting. The ZnO nanorod arrays were first synthesized on the ITO glass and served as the template. Afterward, the ZnO nanorod arrays were converted into CuO nanorod arrays by a facile gas-phase cation exchange process. The CuO nanorod arrays exhibit a promising performance as the efficient photoelectrochemical water splitting electrode with a high photocurrent of  $2.42 \text{ mA cm}^{-2}$  at  $0 \text{ V vs. RHE}$ . Moreover, the successful synthesis of one-dimensional CuO nanostructures in our work provides a new promising platform for new viable nanodevices.

## 2. Experimental

### 2.1. Materials synthesis

ZnO nanorod arrays were synthesized by a hydrothermal method. A thin layer of ZnO was first deposited on the ITO glass using RF magnetron sputtering. The substrate was then kept in an aqueous solution containing  $40 \text{ mM Zn(NO}_3)_2$  and  $40 \text{ mM}$  hexamethylenetetramine under  $90 \text{ }^\circ\text{C}$  for 6 hours. The gas-phase cation exchange was performed to convert the ZnO nanorod arrays to CuO nanorod arrays. In a typical reaction, the as-synthesized ZnO nanorod arrays were placed in a tube furnace and  $0.5 \text{ g}$  of  $\text{CuCl}_2$  was placed upstream at a distance of

$5 \text{ cm}$ . The furnace was heated to  $450 \text{ }^\circ\text{C}$  at  $1 \text{ }^\circ\text{C min}^{-1}$  and kept for 5 minutes under  $50 \text{ sccm Ar}$  flow (ESI, Fig. S1†).

### 2.2. Materials characterization

Field-emission scanning electron microscopy (FESEM) and transmission electron microscopy (TEM) images were obtained on a Hitachi S-4800 FESEM and a JEOL 2010 TEM, respectively. The HRTEM image was captured using a FEI Tecnai G2 F30 Twin TEM. The XRD, Raman, and UV-vis measurements were performed by Rigaku MiniFlex II X-ray diffractometer, Thermo Fisher DXR dispersive Raman spectrometer, and Varian Cary 500 Scan UV-vis NIR spectrophotometer, respectively.

### 2.3. PEC water splitting measurements

A Gamry Reference 600 potentiostat was used for the photoelectrochemical measurements. The photoelectrochemical performance was studied under the back-side simulated AM 1.5 G illumination using a New Port 67005 Arc Lamp. The electrochemical impedance spectroscopy (EIS) measurements were conducted with a potential amplitude of  $5 \text{ mV}$  in the frequency range of  $0.01 \text{ Hz}$  to  $100 \text{ kHz}$ . All the photoelectrochemical measurements were performed in a  $1 \text{ M Na}_2\text{SO}_4$  aqueous electrolyte at ambient temperature.

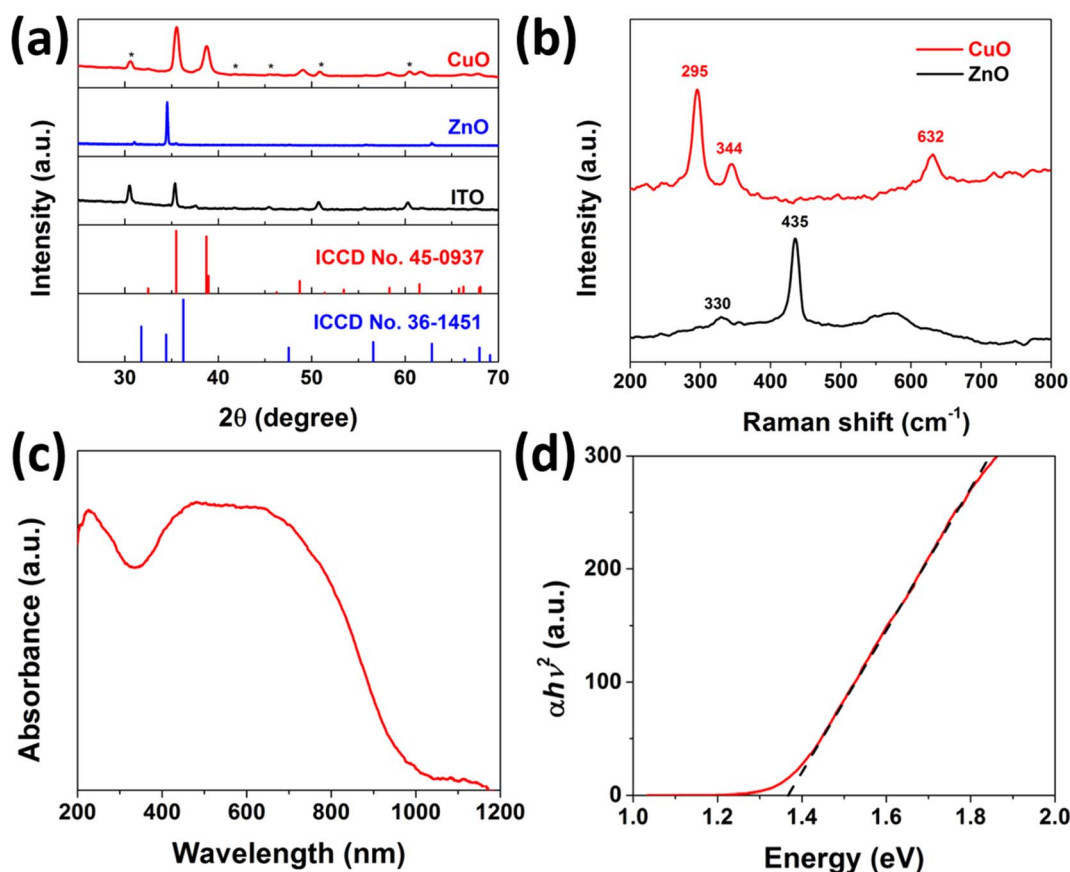


Fig. 1 (a) XRD spectra of ITO glass, ZnO nanorod arrays and CuO nanorod arrays; (b) Raman spectra of ZnO and CuO nanorod arrays; (c) UV-vis absorbance spectrum and (d) Tauc plot of CuO nanorod arrays.



### 3. Results and discussion

The synthesis of CuO nanorod arrays was accomplished by a gas-phase cation exchange reaction using ZnO nanorod arrays as the sacrificial template. The ZnO nanorod arrays were grown on ITO glass by a hydrothermal method. As shown in Fig. S2 in the ESI,<sup>†</sup> the color of the ZnO nanorod arrays on ITO glass changed from gray to black after the cation exchange reaction, which corresponds to the color of ZnO and CuO, respectively. The structure of CuO nanorod arrays was first examined by XRD and Raman. Fig. 1a presents the XRD results of the bare ITO glass and the ZnO nanorod arrays before and after the cation exchange reaction. The XRD spectrum of the pristine ZnO nanorod arrays displays a typical XRD pattern of ZnO with a high-purity wurtzite hexagonal phase (JCPDS card no. 36-1451). After the cation exchange reaction, all of the diffraction peaks, except the peaks from the ITO glass as denoted by asterisks, can be indexed as a monoclinic CuO structure (JCPDS no. 45-0934) with no impurity peaks, such as Cu or Cu<sub>2</sub>O, suggesting the full conversion of ZnO to CuO. Fig. 1b presents the Raman spectra of the ZnO and CuO nanorod arrays. Two peaks located at 435 and 330 cm<sup>-1</sup> for ZnO, which are corresponding to the E<sub>2</sub> mode and the multiple-phonon scattering process, respectively.<sup>25</sup> After the cation exchange reaction, three distinct Raman peaks located at 295, 344, and 632 cm<sup>-1</sup> emerged, which can be attributed to A<sub>g</sub> and two B<sub>g</sub> modes of CuO; no peaks from Cu<sub>2</sub>O can be observed.<sup>26</sup> Both XRD and Raman results prove the

successful transition of ZnO to phase-pure of CuO. The UV-vis absorbance spectrum as shown in Fig. 1c reveals a broad absorption up to 800 nm with an absorption tail that expands to nearly 1200 nm. The bandgap of the CuO nanorod arrays was determined to be 1.37 eV according to the corresponding Tauc plot (Fig. 1d), indicating a good absorption of the solar light in the visible region and great promise for PEC water splitting application.

The morphology and structure of typical CuO nanorods were further investigated using FESEM and TEM. The SEM image of the ZnO nanorod arrays as shown in Fig. 2a reveals the uniform growth of ZnO nanorods on the ITO glass with quasi-vertical alignment. The inset image reveals the typical hexagonal morphology due to its wurtzite crystal structure, with the diameter in the range of 100–200 nm. The average length of ZnO nanorods was measured to be around 2 μm by the cross-section SEM image (Fig. S3a<sup>†</sup>). After the cation exchange reaction, the overall one-dimensional nanorod array structure was well preserved, and the density and the size of the nanorods remain almost unchanged, as shown in Fig. 2b and S3b.<sup>†</sup> By comparison, as shown in the inset high-magnification SEM images before and after the cation exchange reaction, the surface of the nanorods became rough, which is similar to previously reported results on other metal oxides.<sup>20</sup> The TEM image and the HRTEM of the ZnO nanorod were presented in Fig. S4.<sup>†</sup> The ZnO nanorod has a typical diameter of around 100 nm with a smooth surface. The as-synthesized ZnO nanorod exhibits a well-

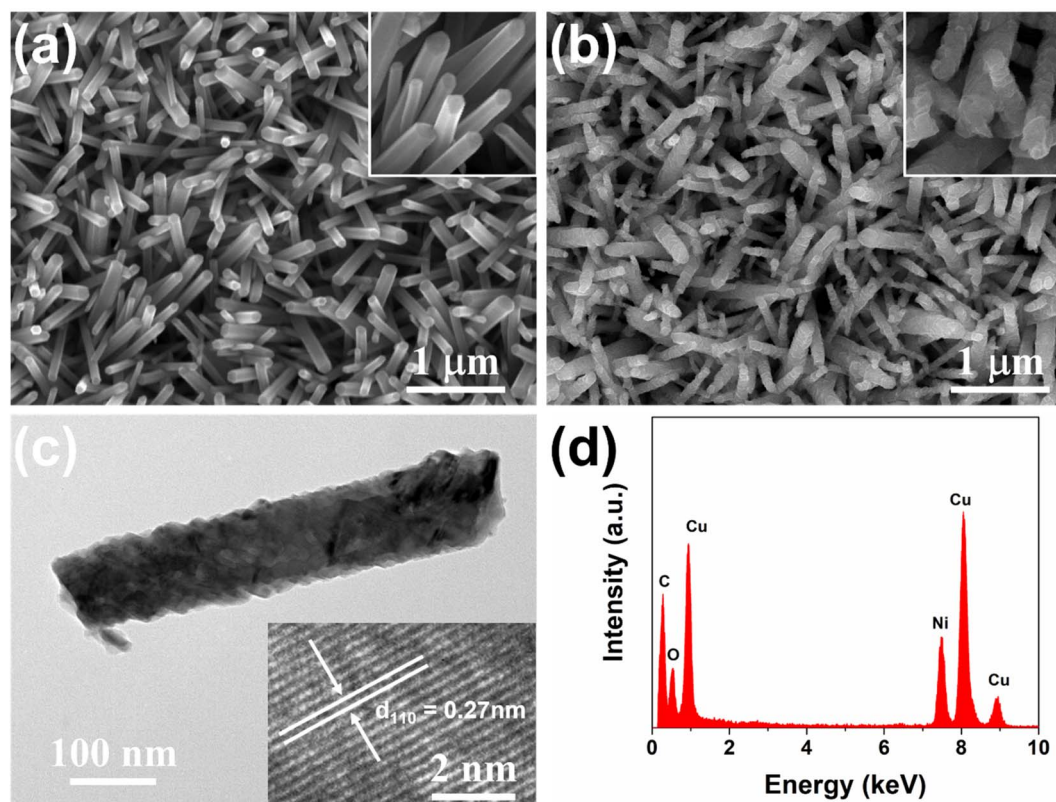


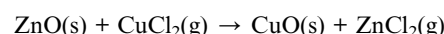
Fig. 2 SEM images of (a) ZnO and (b) CuO nanorod arrays obtained under 450 °C; (c) TEM image of a single CuO nanorod, the inset showing the HRTEM lattice image of the (110) plane; (d) the EDS spectrum of the CuO nanorod, the Ni peak came from the TEM grid.



crystalline nature, with a lattice fringe space of 5.2 Å corresponding to the (0001) plane of the wurtzite ZnO. Fig. 2c presents the TEM image of one typical CuO nanorod with a width of around 100 nm and a rough surface, consistent with the SEM results. Similar to the previous reports, the nanorod became porous after the cation exchange reaction.<sup>20</sup> The inset HRTEM lattice image in Fig. 2c shows the lattice fringes with an inter-plane spacing of 0.27 nm, corresponding to the (110) planes of CuO, revealing the good preservation of the highly crystalline nature of ZnO. Additionally, the EDS spectrum as shown in Fig. 2d presents only Cu and O singles with no Zn peaks that can be found, further confirming the successful synthesis of CuO nanorods.

The cation exchange process was further investigated at different temperatures to understand the growth mechanism. As shown in the XRD spectrum of different reaction temperatures in Fig. 3, the transformation of ZnO into CuO occurs at around 300 °C, resulting in a partial cation exchange product. The ZnO was fully converted to CuO at the temperature range from 350 to 600 °C. The conversion of ZnO into CuO is feasible due to the high vapor pressure of CuCl<sub>2</sub> as the reaction temperature is close to the melting point of CuCl<sub>2</sub>. To our surprise, a further increase of the temperature up to around 650 °C results in the formation of CuO/Cu<sub>2</sub>O and Cu<sub>2</sub>O at 700 °C, which is contrary to the behavior of CuO synthesized by the

thermal oxidation method, where Cu<sub>2</sub>O was first formed at the lower temperature and further oxidized into CuO nanowires.<sup>16</sup> However, the excess temperature causes deterioration to the nanorod structures, as shown in Fig. S5.† Fig. S6† shows the TEM of the nanorod after the cation exchange under 300 °C. A clear interface can be observed as marked using the dashed line, which indicating a core/shell structure was formed. Owing to the fact that the evolution of the CuO was similar to that of previously reported CoO/Co<sub>3</sub>O<sub>4</sub> and Mn<sub>3</sub>O<sub>4</sub> systems, it is reasonable to believe that the translation of ZnO to CuO was realized by the cation exchange process of the outward diffusion of Zn<sup>2+</sup> cations and the inward diffusion of Cu<sup>2+</sup> cations when the thermal disturbance promotes the vibration of surrounding O<sup>2-</sup> anions and further phase conversion by the cation inter-diffusion through interstitial channels.<sup>19,20,27</sup> During the cation exchange process, the Zn species might be pumped away in the form of vaporized ZnCl<sub>2</sub> by the following reaction:<sup>28,29</sup>



The as-synthesized CuO nanorod arrays on ITO glass were unitized as photoelectrode for PEC water splitting to evaluate their practical application. The electrochemical measurements were conducted in a typical three-electrode configuration with an intensity of 100 mW cm<sup>-2</sup> AM 1.5 solar irradiation in a 1 M Na<sub>2</sub>SO<sub>4</sub> electrolyte using the CuO nanorod arrays with the size of 1 × 1 cm<sup>2</sup> as the working electrode, Pt as the counter electrode, and Ag/AgCl as the reference electrode. The linear sweep voltammetry curves of CuO nanorod arrays synthesized under different reaction temperatures are presented in Fig. 4a. All the samples exhibit the typical p-type semiconductor character, as the onset potential shifted from ~200 to ~400 mV vs. RHE when the light was on. As we expected, the photocurrent increased significantly compared to that without light illumination. The optimized reaction temperature was found to be 450 °C, as the photocurrent at 0 V vs. RHE reaches 2.1 mA cm<sup>-2</sup>. By comparison of the TEM images (Fig. 2c and S7†), it is clear that CuO nanorods under different temperatures exhibit distinct surface features. At low temperatures of 400 °C, the CuO nanorod has a relatively smooth surface. As the temperature increases, the surface became rough, which may provide more active sites for the photoelectrochemical, resulting in higher PEC performance. However, as the temperature further increases, the nanostructure of the CuO nanorod was deteriorated, thus affecting the PEC performance. The tradeoff between surface active sites and nanostructure integrity might result in an optimal reaction temperature of 450 °C. To understand the impact of the cation exchange temperature on the PEC performance, electrochemical impedance spectroscopy (EIS) was performed to study the semiconducting properties and charge transport behavior. The Nyquist plots and their corresponding equivalent circuit are shown in Fig. 4b. The measured impedance spectra match well with the equivalent circuit, and the equivalent series resistance *R<sub>S</sub>* and the charge transfer resistance *R<sub>CT</sub>* were presented in the ESI as Table S1.† The *R<sub>CT</sub>*, which is associated with the charge transfer between the CuO electrode and the Na<sub>2</sub>SO<sub>4</sub> electrolyte, was found to play a crucial role

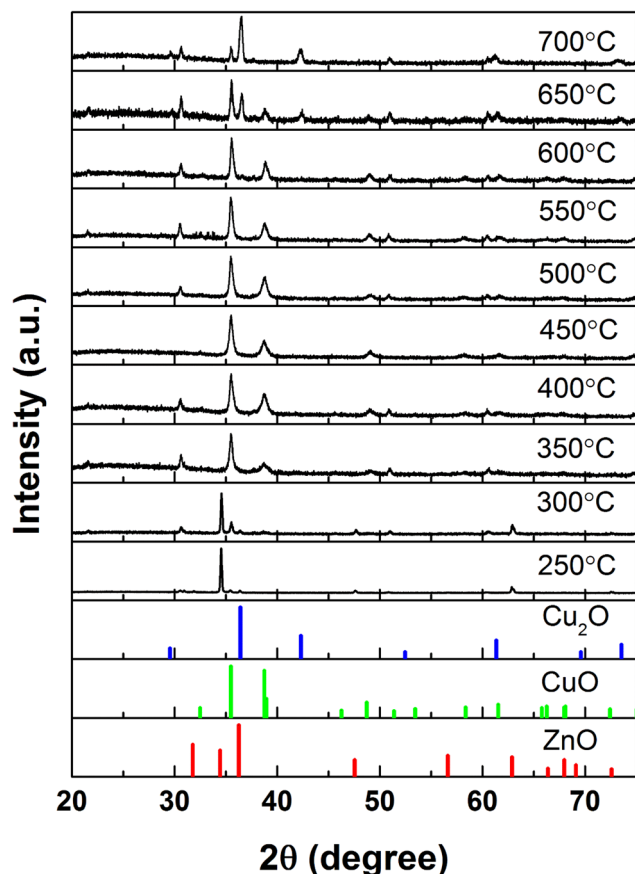


Fig. 3 The XRD spectrum of the gas-phase cation exchange products under different reaction temperatures.



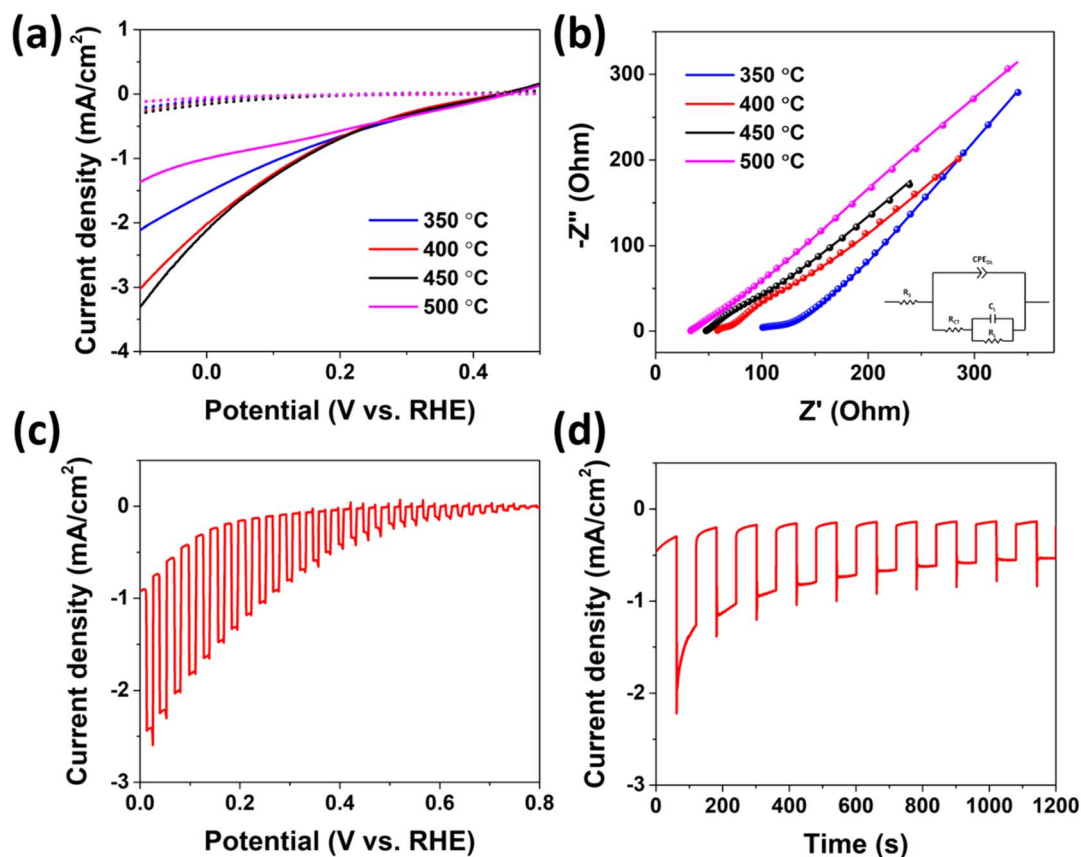


Fig. 4 (a) Current–potential characteristics and (b) EIS spectrum of the CuO nanorod arrays synthesized under different reaction temperatures. (c) Current–potential characteristics and (d) stability test of CuO nanorod arrays synthesized under 450 °C.

in terms of affecting the PEC performance, as the photocurrent is proportional to the value of  $R_{CT}$  measured. Under the optimized reaction temperature of 450 °C, the charge transfer from CuO to the electrolyte was facilitated, which leads to the enhancement of the conductivity and the increase of the photocurrent.<sup>4</sup> The photoelectrochemical behaviors of the optimized CuO electrode were further investigated by the linear sweep voltammetry under a chopped light illumination as shown in Fig. 4c. The overall  $i$ - $v$  behavior is similar to that of the steady-state light illumination, though the increment of the dark current was observed, especially towards more negative

potential. This was attributed to the possible shunt of the device caused by the exposure of the substrate.<sup>12</sup> Nevertheless, the photocurrent reaches  $2.42 \text{ mA cm}^{-2}$  at 0 V vs. RHE, which is among the highest value in the pure CuO system reported (Table 1). One main issue of CuO for PEC water splitting applications is its poor stability in aqueous electrolytes due to the photocorrosion of CuO into metallic Cu since the decomposition potential of copper oxide lies close to or within the bandgap.<sup>30–32</sup> As shown in Fig. 4d, the photostability measurement was performed at a fixed potential of 0 V vs. RHE. The photocurrent declines at a very short time but stabilizes

Table 1 Comparison of the PEC performance of various CuO-based electrodes

PEC electrode	Substrate	Synthesis method	Photocurrent density	Ref.
CuO nanorod arrays	ITO glass	Gas-phase cation exchange method	$2.42 \text{ mA cm}^{-2}$ at 0 V vs. RHE	Present work
CuO nanowires	Cu foil	Thermal oxidation	$1.4 \text{ mA cm}^{-2}$ at 0 V vs. RHE	15
CuO nanoleaves	FTO glass	Aqueous solution growth	$1.5 \text{ mA cm}^{-2}$ at 0 V vs. RHE	33
Nanostructured CuO	Cu foil	Chemical bath deposition (CBD)	$1.3 \text{ mA cm}^{-2}$ at 0 V vs. RHE	34
Tree branch-shaped CuO	FTO glass	Hybrid microwave annealing (HMA)	$4.4 \text{ mA cm}^{-2}$ at 0 V vs. RHE	35
CuO thin films	FTO glass	Sputtering	$1.68 \text{ mA cm}^{-2}$ at 0 V vs. RHE	36
CuO nanoparticles	FTO glass	Sol-gel dip-coating process	$0.94 \text{ mA cm}^{-2}$ at 0 V vs. RHE	37
Nanotextured CuO films	ITO glass	Cold spray method	$3.1 \text{ mA cm}^{-2}$ at 0 V vs. RHE	38
CuO thin films	FTO glass	Microwave-assisted method	$1.15 \text{ mA cm}^{-2}$ at 0 V vs. RHE	39



afterward, resulting in reasonable stability by retaining about 50% of the initial photocurrent after 1200 s under the light on/off condition.

## 4. Conclusion

In summary, we have demonstrated a gas-phase cation exchange method to synthesize CuO nanorod arrays by using ZnO as the sacrificial template. The optimized CuO nanorod arrays exhibit excellent performance with a photocurrent density up to 2.42 mA cm<sup>-2</sup> at 0 V vs. RHE. This work provides a new pathway for the efficient synthesis of one-dimensional CuO nanostructures and demonstrated the feasibility on the practical application of photoelectrochemical water splitting.

## Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.

## Conflicts of interest

The authors have no conflicts to disclose.

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## References

- 1 Y.-H. Chiu, T.-H. Lai, M.-Y. Kuo, P.-Y. Hsieh and Y.-J. Hsu, *APL Mater.*, 2019, **7**, 080901.
- 2 P.-Y. Hsieh, J.-Y. Wu, T.-F. M. Chang, C.-Y. Chen, M. Sone and Y.-J. Hsu, *Arab. J. Chem.*, 2020, **13**, 8372–8387.
- 3 C.-W. Tsao, M.-J. Fang and Y.-J. Hsu, *Coord. Chem. Rev.*, 2021, **438**, 213876.
- 4 S. Masudy-Panah, R. S. Moakhar, C. S. Chua, H. R. Tan, T. I. Wong, D. Chi and G. K. Dalapati, *ACS Appl. Mater. Interfaces*, 2016, **8**, 1206–1213.
- 5 R. S. Moakhar, S. M. Hosseini-Hosseinabad, S. Masudy-Panah, A. Seza, M. Jalali, H. Fallah-Arani, F. Dabir, S. Gholipour, Y. Abdi, M. Bagheri-Hariri, N. Riahi-Noori, Y.-F. Lim, A. Hagfeldt and M. Saliba, *Adv. Mater.*, 2021, **33**, 2007285.
- 6 A. Paracchino, V. Laporte, K. Sivula, M. Grätzel and E. Thimsen, *Nat. Mater.*, 2011, **10**, 456–461.
- 7 Y.-F. Lim, C. S. Chua, C. J. J. Lee and D. Chi, *Phys. Chem. Chem. Phys.*, 2014, **16**, 25928–25934.
- 8 Z. Zhang, R. Dua, L. Zhang, H. Zhu, H. Zhang and P. Wang, *ACS Nano*, 2013, **7**, 1709–1717.
- 9 W. Yang, R. R. Prabhakar, J. Tan, S. D. Tilley and J. Moon, *Chem. Soc. Rev.*, 2019, **48**, 4979–5015.
- 10 C. Cai, S. Han, X. Zhang, J. Yu, X. Xiang, J. Yang, L. Qiao, X. Zu, Y. Chen and S. Li, *RSC Adv.*, 2022, **12**, 6205–6213.
- 11 M.-J. Fang, C.-W. Tsao and Y.-J. Hsu, *J. Phys. D: Appl. Phys.*, 2020, **53**, 143001.
- 12 J. Luo, L. Steier, M.-K. Son, M. Schreier, M. T. Mayer and M. Grätzel, *Nano Lett.*, 2016, **16**, 1848–1857.
- 13 L. Xiang, J. Guo, C. Wu, M. Cai, X. Zhou and N. Zhang, *J. Mater. Res.*, 2018, **33**, 2264–2280.
- 14 X. Zhao, P. Wang, Z. Yan and N. Ren, *Chem. Phys. Lett.*, 2014, **609**, 59–64.
- 15 J. Li, X. Jin, R. Li, Y. Zhao, X. Wang, X. Liu and H. Jiao, *Appl. Catal., B*, 2019, **240**, 1–8.
- 16 X. Jiang, T. Herricks and Y. Xia, *Nano Lett.*, 2002, **2**, 1333–1338.
- 17 X. Gao, J. Li, R. Du, J. Zhou, M.-Y. Huang, R. Liu, J. Li, Z. Xie, L.-Z. Wu, Z. Liu and J. Zhang, *Adv. Mater.*, 2017, **29**, 1605308.
- 18 Q. Zhang, K. Zhang, D. Xu, G. Yang, H. Huang, F. Nie, C. Liu and S. Yang, *Prog. Mater. Sci.*, 2014, **60**, 208–337.
- 19 C. W. Na, S.-Y. Park, J.-H. Chung and J.-H. Lee, *ACS Appl. Mater. Interfaces*, 2012, **4**, 6565–6572.
- 20 H. Zhang, T. Ling and X. W. Du, *Chem. Mater.*, 2015, **27**, 352–357.
- 21 T. Zhang, M.-Y. Y. Wu, D.-Y. Y. Yan, J. Mao, H. Liu, W.-B. B. Hu, X.-W. W. Du, T. Ling and S.-Z. Z. Qiao, *Nano Energy*, 2018, **43**, 103–109.
- 22 T. Ling, T. Zhang, B. Ge, L. Han, L. Zheng, F. Lin, Z. Xu, W. W.-B. Hu, X.-W. X. Du, K. Davey and S.-Z. S. Qiao, *Adv. Mater.*, 2019, **31**, 1807771.
- 23 Y.-J. Li, L. Cui, P.-F. Da, K.-W. Qiu, W.-J. Qin, W.-B. Hu, X.-W. Du, K. Davey, T. Ling and S.-Z. Qiao, *Adv. Mater.*, 2018, **30**, 1804653.
- 24 T. Ling, P. Da, X. Zheng, B. Ge, Z. Hu, M. Wu, X.-W. Du, W.-B. Hu, M. Jaroniec and S.-Z. Qiao, *Sci. Adv.*, 2018, **4**, eaau6261.
- 25 K. Wang, J. Chen, W. Zhou, Y. Zhang, Y. Yan, J. Pern and A. Mascarenhas, *Adv. Mater.*, 2008, **20**, 3248–3253.
- 26 J. F. Xu, W. Ji, Z. X. Shen, W. S. Li, S. H. Tang, X. R. Ye, D. Z. Jia and X. Q. Xin, *J. Raman Spectrosc.*, 1999, **30**, 413–415.
- 27 C. W. Na, H.-S. Woo, H.-J. Kim, U. Jeong, J.-H. Chung and J.-H. Lee, *CrystEngComm*, 2012, **14**, 3737.
- 28 C. W. W. Na, S.-Y. Park, J.-H. Chung and J.-H. Lee, *ACS Appl. Mater. Interfaces*, 2012, **4**, 6565–6572.
- 29 E. Thimsen, Q. Peng, A. B. F. Martinson, M. J. Pellin and J. W. Elam, *Chem. Mater.*, 2011, **23**, 4411–4413.
- 30 U. Shaislamov, K. Krishnamoorthy, S. J. Kim, A. Abidov, B. Allabergenov, S. Kim, S. Choi, R. Suresh, W. M. Ahmed and H.-J. Lee, *Int. J. Hydrogen Energy*, 2016, **41**, 2253–2262.
- 31 C. Y. Toe, Z. Zheng, H. Wu, J. Scott, R. Amal and Y. H. Ng, *Angew. Chem., Int. Ed.*, 2018, **57**, 13613–13617.
- 32 H. Xing, L. E. Z. Guo, D. Zhao, X. Li and Z. Liu, *Inorg. Chem. Front.*, 2019, **6**, 2488–2499.
- 33 A. Kushwaha, R. S. Moakhar, G. K. L. Goh and G. K. Dalapati, *J. Photochem. Photobiol., A*, 2017, **337**, 54–61.



## Paper

- 34 A. Ray, I. Mukhopadhyay, R. Pati, Y. Hattori, U. Prakash, Y. Ishii and S. Kawasaki, *J. Alloys Compd.*, 2017, **695**, 3655–3665.
- 35 Y. J. Jang, J. W. Jang, S. H. Choi, J. Y. Kim, J. H. Kim, D. H. Youn, W. Y. Kim, S. Han and J. S. Lee, *Nanoscale*, 2015, **7**, 7624–7631.
- 36 S. Masudy-Panah, R. S. Moakhar, C. S. Chua, A. Kushwaha, T. I. Wong and G. K. Dalapati, *RSC Adv.*, 2016, **6**, 29383–29390.
- 37 J. Toupin, H. Strubb, S. Kressman, V. Artero, N. Krins and C. Laberty-Robert, *J. Sol-Gel Sci. Technol.*, 2019, **89**, 255–263.
- 38 J. G. Lee, D.-Y. Kim, J.-H. Lee, M. Kim, S. An, H. S. Jo, C. Nervi, S. S. Al-Deyab, M. T. Swihart and S. S. Yoon, *ACS Appl. Mater. Interfaces*, 2016, **8**, 15406–15414.
- 39 S. M. Hosseini H., R. S. Moakhar, F. Soleimani, A. Goudarzi and S. K. Sadrnezhad, *ECS Trans.*, 2020, **97**, 845.

