


 Cite this: *RSC Adv.*, 2023, **13**, 8706

Petal-like Mn-doped α -Ni(OH)₂ nanosheets for high-performance Li–S cathode material†

 Changfeng Zhao,^a Hanyang Liu,^a Jiawei Liu,^b Yanhong Shi,^a Shuguang Wang,^a Qiwei Tang,^a Xiangbing Zhu,^a Huimin Zhang^a and Yan Zhao^a

Lithium–sulphur (Li–S) batteries are high-energy-density and cost-effective batteries. Herein, petal-like Ni_{1–x}Mn_x(OH)₂ ($x \approx 0.04$) nanosheets were synthesised using a hydrothermal method and the electrical conductivity of Ni(OH)₂ was improved by applying the cathode functional materials in Li–S batteries. With up to 5 mg cm^{–2} of S content in the cathode, the fabricated Ni_{1–x}Mn_x(OH)₂ electrode exhibited specific discharge capacities up to 1375 and 1150 mA h g^{–1} at 0.2 and 0.5C, and retained this capacity at 813 and 714 mA h g^{–1} after 200 cycles, respectively. Electrochemical measurement results show that Ni_{1–x}Mn_x(OH)₂ plays a critical role in Li–S batteries as it has a larger specific surface area than Ni(OH)₂, which has superior adsorption performance toward lithium polysulphides. Moreover, the conductivity performance of Ni_{1–x}Mn_x(OH)₂ is significantly better than that of Ni(OH)₂, which improves the electrochemical reaction kinetics of the Li–S batteries.

 Received 3rd January 2023
 Accepted 10th March 2023

DOI: 10.1039/d3ra00032j

rsc.li/rsc-advances

1. Introduction

With the rapidly increasing demand for energy applications in various fields, such as stationary storage, military power supplies and transportation facilities, the energy densities of commercial batteries (e.g. Ni–H [80 W h kg^{–1}] and Li-ion [300 W h kg^{–1}] batteries) cannot fulfil the current energy requirements of industries. Therefore, batteries with low costs and high energy densities/specific capacities are in critical demand for energy storage. In this regard, lithium–sulphur (Li–S) batteries have some obvious advantages, such as the low cost of S, high theoretical specific energy (1672 mA h g^{–1}) and environmental friendliness; such characteristics have significantly promoted the high-speed development of Li–S batteries.^{1–3} However, the following defects have hampered the application of Li–S batteries: (1) the large volume expansion (80%) that occurs in cathodes when the conversion reaction occurs; (2) the shuttle effect of Li polysulphides (LiPSs); and (3) the electrical insulation nature of S and its process products (Li₂S₂/Li₂S).^{4–6} Commendable progress has been made in the study of both modified separators and cathodes to ameliorate the lifespans and energy density of Li–S batteries.

Because of the abundant surface hydroxyl groups of polar metal hydroxides such as Ca(OH)₂ (ref. 7) and Ni(OH)₂,^{8,9} polysulphides exhibit strong interactions. For instance, nickel

hydroxide nanosheets are important encapsulation materials because of the high chemical adsorption of polysulphides; moreover, the size of the nickel hydroxide nanosheets blocks polysulphides but allows Li⁺ to pass through.^{8,9} Ni(OH)₂ materials, however, have some disadvantages as well: poor conductivity and less electrochemically active areas that confine the kinetic diffusion to nickel hydroxide materials. Thus, the component and structure of nickel hydroxide requires further optimisation to address these issues. Furthermore, conductive materials with large specific surface areas compounded with Ni(OH)₂ materials can enhance band structures and charge transfers, exposing catalytic active sites and avoiding nanosheet aggregation.^{10,11} Another effective method of promoting the physicochemical property of Ni(OH)₂ materials is heterostructure or atomic doping. The conductivity of Ni(OH)₂ will significantly improve when cobalt (Co), manganese (Mn), iron (Fe) or aluminium (Al) atomic dopes in Ni(OH)₂ materials.^{12,13}

However, different atomic dopes in Ni(OH)₂ materials present different layered double hydroxides (LDHs) such as NiCo-LDH and NiMn-LDH.^{14–17} The thin layer structures of LDHs increase their defects and surface areas, which can change their physicochemical properties and electronic structures. Of these materials, NiMn-LDH is a nontoxic, low-cost and highly active material, and the abovementioned properties make the use of Li–S batteries profitable.

Under this context, we developed a cost-effective and simple method for synthesising Ni_{1–x}Mn_x(OH)₂ ($x \approx 0.04$) nanowalls with highly uniform and large areas. With carbon nanotube (CNT), Ni_{1–x}Mn_x(OH)₂ and S (mass ratio of 15 : 10 : 75) content in the cathode, the first discharge specific capacity reached up to 1375 mA h g^{–1} at 0.2C, and still reached 813 mA h g^{–1} after

^aSchool of Energy and Mechanical Engineering, Dezhou University, DeZhou, Shandong, 253023, P. R. China. E-mail: 867713584@qq.com

^bCollege of Agriculture, Shihezi University, Shihezi, Xinjiang, 832003, P. R. China

 † Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3ra00032j>


200 cycles. The above-described cell exhibits excellent rate performance. These superior electrochemical properties evidence that the conductivity and electrochemical reaction kinetics of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ were effectively enhanced after Mn^{2+} was doped into $\text{Ni}(\text{OH})_2$. These enhanced properties of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ can be ascribed to the following two reasons: first, $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ has an oversized specific surface area and superior adsorption performance for LiPSs, which can inhibit the shuttle effect of soluble LiPSs to a certain extent during discharge. Second, the conductivity performance of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ is significantly improved, which accelerates the electrochemical reaction kinetics of the charge–discharge process.

2. Experimental

2.1 Material synthesis

The Mn-doped $\text{Ni}(\text{OH})_2$ material was fabricated quickly *via* the hydrothermal method. To this end, 50 mg of $\text{Mn}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$, 1105 mg of $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (Mn : Ni atomic ratio of 5 : 95) and 560 mg of hexamethylenetetramine (HMTA) were dispersed in 40 mL of deionised water. The mixtures were subjected to magnetic stirring for 30 min and then placed in a three-necked flask for hydrothermal treatment at 95 °C for 6 h. Oxygen was removed from the flask using Ar, and both sides of the flask were sealed. Notably, the third bottle mouth was sealed with a balloon to relieve the pressure in the flask during heating. The obtained light green sample was thoroughly washed with deionised water several times. The sample was dried at 60 °C for 12 h in an oven and defined as $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$. For comparison, the atomic ratio of Mn : Ni was changed to 1 : 9, and the sample thus obtained was defined as NiMn-LDH. $\text{Ni}(\text{OH})_2$ synthesis was similar to $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ synthesis with the exception that the solution contained no Mn^{2+} . Finally, 1000 mg of $\text{Mn}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$, and 560 mg of HMTA were reacted in the same manner as that described above.

2.2 Cathode fabrication

The weight ratio of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ (or $\text{Ni}(\text{OH})_2$), S and CNT is 10 : 75 : 15, or the weight ratio of S and CNT is 75 : 25. Then, the different mixtures were placed in different polytetrafluoroethylene reactors after they were fully ground and then heated at 155 °C for 12 h. The weight ratio of the composite (CNT/S/ $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ or CNT/S/ $\text{Ni}(\text{OH})_2$ or CNT/S) prepared above, super-P (SP), polyvinylidene fluoride (PVDF) is 80 : 10 : 10, the mixture was milled with *N*-methyl-2-pyrrolidone (NMP) to form uniform slurries. These slurries were pasted onto an aluminium foil each with a thickness of 300 μm and dried at 60 °C for 12 h in an oven with N_2 . Next, the dried slurries were cut into 12 mm-diameter samples on each of which about 5 mg cm^{-2} of S was coated. The sample discs are denoted as CNT/S/ $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, CNT/S/ $\text{Ni}(\text{OH})_2$ and CNT/S, respectively.

2.3 Symmetric cell testing

For symmetric cell measurements, SP, PVDF and the CNT/ $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ (mass ratio of 3 : 2) composites (or CNT/

$\text{Ni}(\text{OH})_2$, the same mass ratio as above, or CNT) were weighed to the mass ratio of 10 : 10 : 80. The sample discs were prepared *via* the cathode method, and the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ (or $\text{Ni}(\text{OH})_2$ or CNT) loading on the disc was approximately 0.5 mg cm^{-2} . The same discs were used as both the anode and cathode to assemble CR2025 cells. Li_2S_6 (0.2 M, 20 μL) was used as the electrolyte and the Celgard 2400 micromembrane was used as the separator. Cyclic voltammetry (CV) was tested on different symmetric cells configurations with a voltage window of -1.0 to 1.0 V at scanning rate of 3 mV s^{-1} .

2.4 Li_2S nucleation tests

Li metal discs and the prepared sample discs from Section 2.3 were utilised as the anodes and cathodes, respectively. The Celgard 2400 micromembrane was used as a separator. 0.1 M Li_2S_8 electrolyte (20 μL) was dropped onto the cathode and 1 M lithium bis(trifluoromethane sulphonimide) (LiTFSI) electrolyte without Li_2S_8 (20 μL) was dripped onto the anode. All prepared coin cells were firstly discharged at 0.112 mA to 2.06 V and then potentiostatic discharge curves were recorded at 2.05 V until the current was below 10^{-5} A.

2.5 Electrochemical measurements

We used the CNT/S/ $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, CNT/S/ $\text{Ni}(\text{OH})_2$ or CNT/S disc as the cathode, the Celgard 2400 micromembrane as the separator, metallic Li as the anode, 1 wt% LiNO_3 and 1 M LiTFSI dissolved in DME/DOL (1 : 1, by volume) as the electrolyte. The obtained coin cell was denoted as $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, $\text{Ni}(\text{OH})_2$ or CNT. Charge–discharge performances for all three were tested on a Neware cell test analyser with a voltage window of 1.6–2.8 V. Electrochemical impedance spectroscopy (EIS) and CV measurements were performed on a CHI760C electrochemical workstation.

2.6 Material characterisation

The prepared samples were examined through scanning electron microscopy and energy dispersive spectrometer (EDS) (SEM, Zeiss G-500), transmission electron microscopy (TEM, JEOL, JEM-1011), selected-area electron diffraction (SAED) and high-resolution TEM (HRTEM, JEOL, JEM2100Plus), X-ray photoelectron spectroscopy (XPS, ESCALAB 250) and X-ray diffraction (XRD, Bruker D8). Surface areas were calculated using the Brunauer–Emmett–Teller (BET) method. Mass ratios of S and $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ (or $\text{Ni}(\text{OH})_2$) in the sample were pre-processed using a HCl solution and thermogravimetric analysis (TGA) (Netzsch, STA 449 F3, 5 °C min^{-1} in N_2).

3. Results and discussion

A preparation method of petal-like $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ nanosheets is shown in Fig. 1a.

The SEM images in Fig. 1b and S1a† show that the $\text{Ni}(\text{OH})_2$ displayed an ultrathin petal-like sphere with an average diameter of about 5 μm . Some small-size nanosheets were grown in the core of the petal-like sphere, and other large-size nanosheets were wrapped in the small-size nanosheets. The



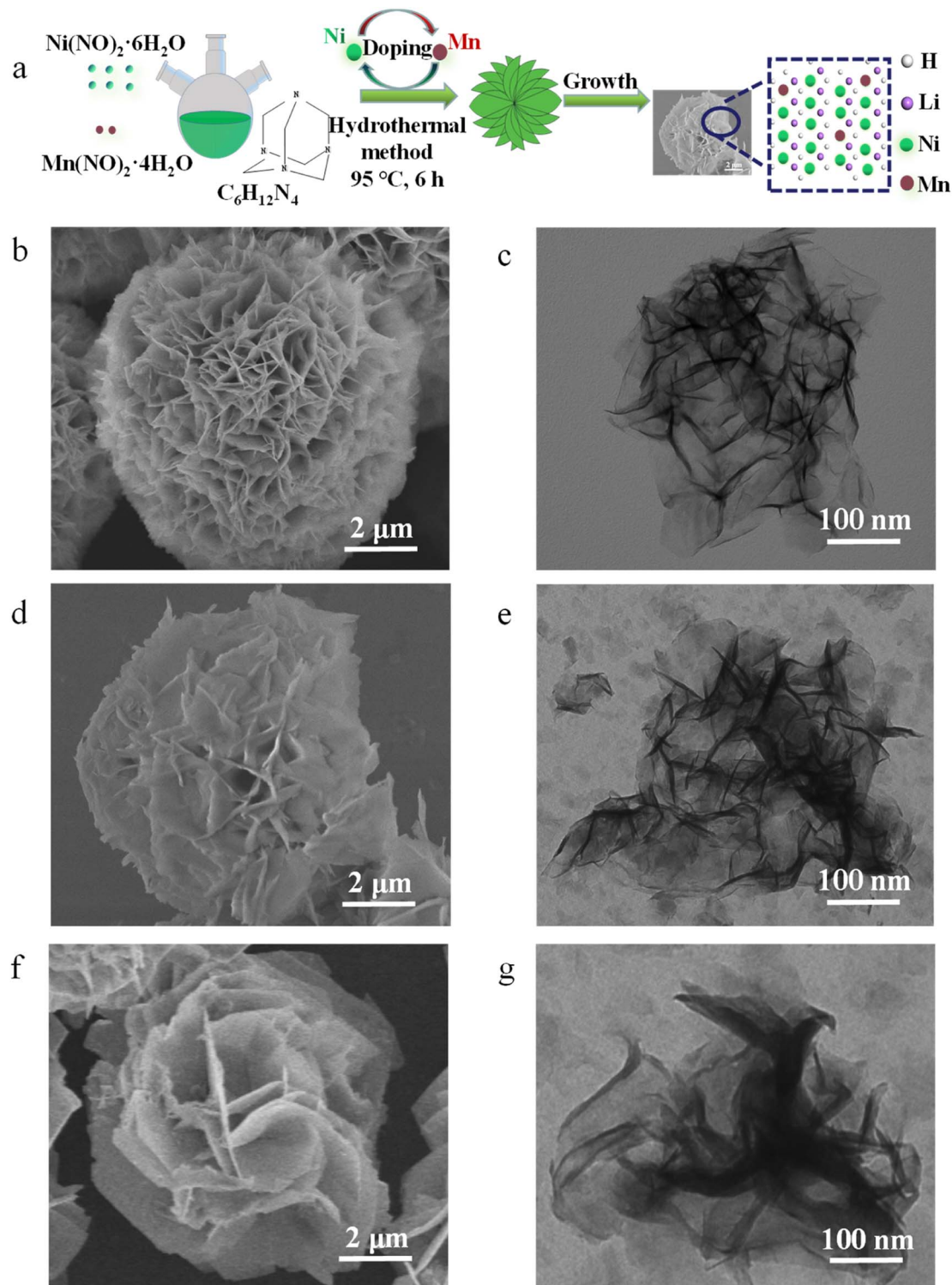


Fig. 1 (a) Preparation method of petal-like $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ nanosheets. SEM images of (b) $\text{Ni}(\text{OH})_2$, (d) $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ and (f) NiMn-LDH . TEM images of (c) $\text{Ni}(\text{OH})_2$, (e) $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ and (g) NiMn-LDH .

thickness of the 'flower-like' nanosheets was about several nanometres, and their size was of several microns. Interestingly, with Mn doping, the nanosheet structure of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ was distinctly rigid, and the thickness of these nanosheets exceeded that of the $\text{Ni}(\text{OH})_2$ nanosheets (Fig. 1d and S1b†).

Such $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ structures provide several active sites for charge transport and electrochemical reaction. With increasing Mn doping, the NiMn-LDH nanosheet became thicker and more scattered than the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ nanosheets (Fig. 1f and S1c†). The excessive Mn ions provided more selectivity as the



growth of the crystal nucleus continued, and some Mn ions might not have been doped into Ni(OH)₂. The TEM image of the Ni(OH)₂ reveal that the thickness of the nanoscale petal-like slices was about 10 nm, and the layered overlay of the nano-sheets was more obvious therein (Fig. 1d). Further, TEM observations reveal that the thickness and shape of Ni_{1-x}Mn_x(OH)₂ (Fig. 1e) and NiMn-LDH (Fig. 1g) were similar to those of Ni(OH)₂. N₂ adsorption/desorption testing was performed to further identify the surface area of Ni_{1-x}Mn_x(OH)₂ and Ni(OH)₂. The adsorption/desorption curves of Ni_{1-x}Mn_x(OH)₂ and Ni(OH)₂ exhibited typical IV isotherms (Fig. S2†). The specific surface area of Ni(OH)₂ was measured to be about 65.25 m² g⁻¹, which is significantly smaller than that of Ni_{1-x}Mn_x(OH)₂ (approximately 84.75 m² g⁻¹). These results prove that Mn doping in Ni(OH)₂ can increase the surface areas of and defects in Ni(OH)₂, significantly increasing the contact area between the electrolyte and electrode materials of Li-S batteries.

The crystallinities and structural characterisations of Ni(OH)₂, Ni_{1-x}Mn_x(OH)₂ and NiMn-LDH were inspected *via* XRD (Fig. 2a). The results show that all the main diffraction peaks of Ni(OH)₂ were ascribable to α -Ni(OH)₂ (JCPDS No. 38-0715), and no other obvious diffraction peaks were observed.^{18,19}

This indicates that α -Ni(OH)₂ was the main species of the above-prepared material. However, the intensity of the diffraction peaks was weaker than that of Ni(OH)₂, indicating that Mn ions have been doped into Ni(OH)₂; this result agrees with the HRTEM scans and SAED patterns for Ni_{1-x}Mn_x(OH)₂. The intensity and position of the main diffraction peaks with the sample (the atomic ratio of Ni:Mn was 90:10) obviously differed from those of α -Ni(OH)₂, and had obvious new diffraction peaks, which agreed with the results obtained for Mn(OH)₂ (JCPDS No. 18-0787).²⁰ The results revealed that the amount of Mn ions doped in Ni(OH)₂ was limited under such hydrothermal reaction conditions. The main diffraction peaks of the sample (which only contained Mn²⁺) corresponded to pure Mn₃O₄ (JCPDS No. 24-0734), revealing that Mn(OH)₂ easily converted to oxides under the presence of oxygen and heating.

The EDS elemental analysis of the sample (Mn:Ni atomic ratio of 5:95) showed that the atomic ratio of Ni:Mn was about 96:4 (Fig. 2b), demonstrating that Ni atoms were replaced by a small amount of Mn atoms in the Ni-Mn hydroxides, defined as Ni_{1-x}Mn_x(OH)₂ ($x \approx 0.04$). The element mapping images of Ni_{1-x}Mn_x(OH)₂ show that the O, Mn and Ni elements were homogeneously distributed across majority of the flower-like

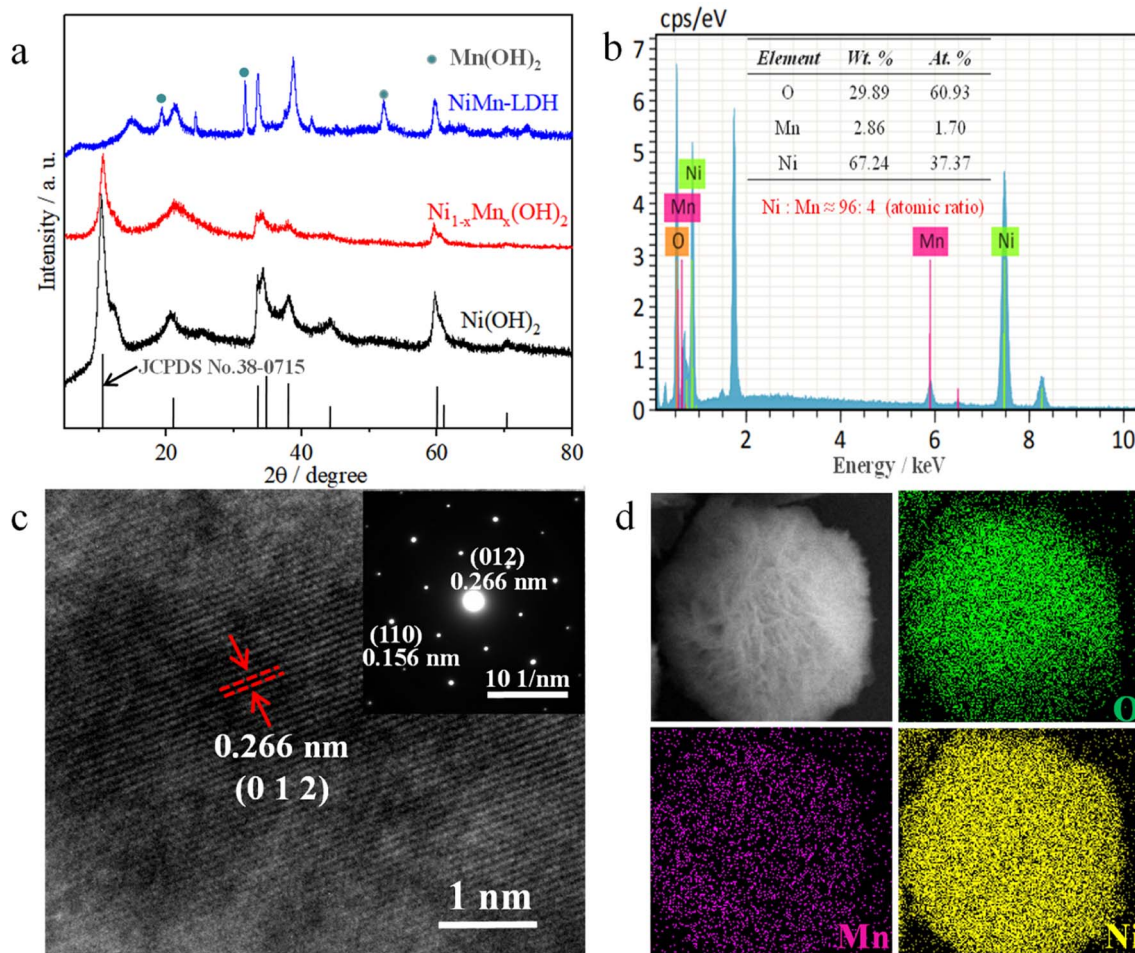


Fig. 2 (a) XRD patterns of NiMn-LDH, Ni_{1-x}Mn_x(OH)₂ and Ni(OH)₂. (b) EDS results of Ni_{1-x}Mn_x(OH)₂. (c) HRTEM images and SAED patterns for Ni_{1-x}Mn_x(OH)₂. (d) Elemental mappings of Mn, O and Ni of Ni_{1-x}Mn_x(OH)₂.



nanosheet area (Fig. 2d). Compared with the Ni signal, the Mn signal was obviously minor in terms of content but was clearly visible. The HRTEM image of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ from Fig. 2c shows a lattice space of 0.266 nm, which matched well with the (0 1 2) plane of $\alpha\text{-Ni}(\text{OH})_2$. The characteristic crystal planes of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ were also matched well with (0 1 2) and (1 1 0) planes of $\alpha\text{-Ni}(\text{OH})_2$ in the SAED pattern.

The corresponding XPS results shown in Fig. 3 indicate the chemical valence states and composition of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$. The survey spectrum indicates the presence of Ni, Mn, O and C elements in the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ sample (Fig. 3a). For the Ni 2p spectra shown in Fig. 3b, two spin-orbit split doublet binding energies at 872.7 and 856.9 eV were matched to Ni 2p_{1/2} and Ni 2p_{3/2}, and the two peaks observed at around 860.9 and 878.8 eV, respectively, were corresponded with the shake-up satellites of Ni 2p (identified as 'Sat.'). The Mn 2p spectra showed two binding energies at 653.9 and 642.7 eV (Fig. 3c), matched to Mn 2p_{1/2} and Mn 2p_{3/2}, respectively. Note that the O 1s peak (Fig. 3d) centred at 531.1 eV corresponded to a hydroxyl group, supporting the existence of Ni and Mn hydroxides.^{21,22}

Fig. S3† shows the Li_2S_6 adsorption tests between $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ and $\text{Ni}(\text{OH})_2$ to further confirm the effect of absorption with polysulphide. Compared to that of the $\text{Ni}(\text{OH})_2$ powder, the colour of Li_2S_6 containing $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ powder changed from dark yellow to transparent after 5 h, indicating that $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ had a more obvious adsorption effect on

Li_2S_6 than $\text{Ni}(\text{OH})_2$. Additionally, the acid immersion treatment and TGA measurement clarified the composition of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, S and CNT in the composite, and Fig. S4a† shows the fraction of S to be approximately 75 wt% and that of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ to be 10 wt%. The composition of $\text{Ni}(\text{OH})_2/\text{S}/\text{CNT}$ indicates the fractions of S to be approximately 74 wt% and $\text{Ni}(\text{OH})_2$ at 10 wt% (Fig. S4b†). The composition of S/CNT indicated the fraction of S to be approximately 75 wt% (Fig. S4c†).

CV were measured to examine the electrochemical activities of various electrodes in cells, wherein the test voltage window was 1.6–2.9 V at 0.1 mV s⁻¹. The $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode displayed reduction peaks at 2.32 and 2.02 V, involving the conversion of S_8 into solution LiPSs and further into $\text{Li}_2\text{S}_2/\text{Li}_2\text{S}$ (Fig. 4a). Conversely, the oxidation peaks at 2.37 and 2.41 V signified the converse process of $\text{Li}_2\text{S}_2/\text{Li}_2\text{S}$ conversion into solution LiPSs and S_8 , and the potential difference (PD) was 0.35 V. The two reduction peaks of the $\text{Ni}(\text{OH})_2$ electrode were observed at 2.30 and 2.03 V, the oxidation peak was located only at 2.51 V and the PD was up to 0.53 V, which is considerably higher than that observed for the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode. The $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode also showed a few increases in current sharpness compared with the $\text{Ni}(\text{OH})_2$ electrode, again confirming that $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ can efficiently accelerate polysulphide conversion with reduced polarisation. The PD of the CNT electrode (0.32 V) was smaller than $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$

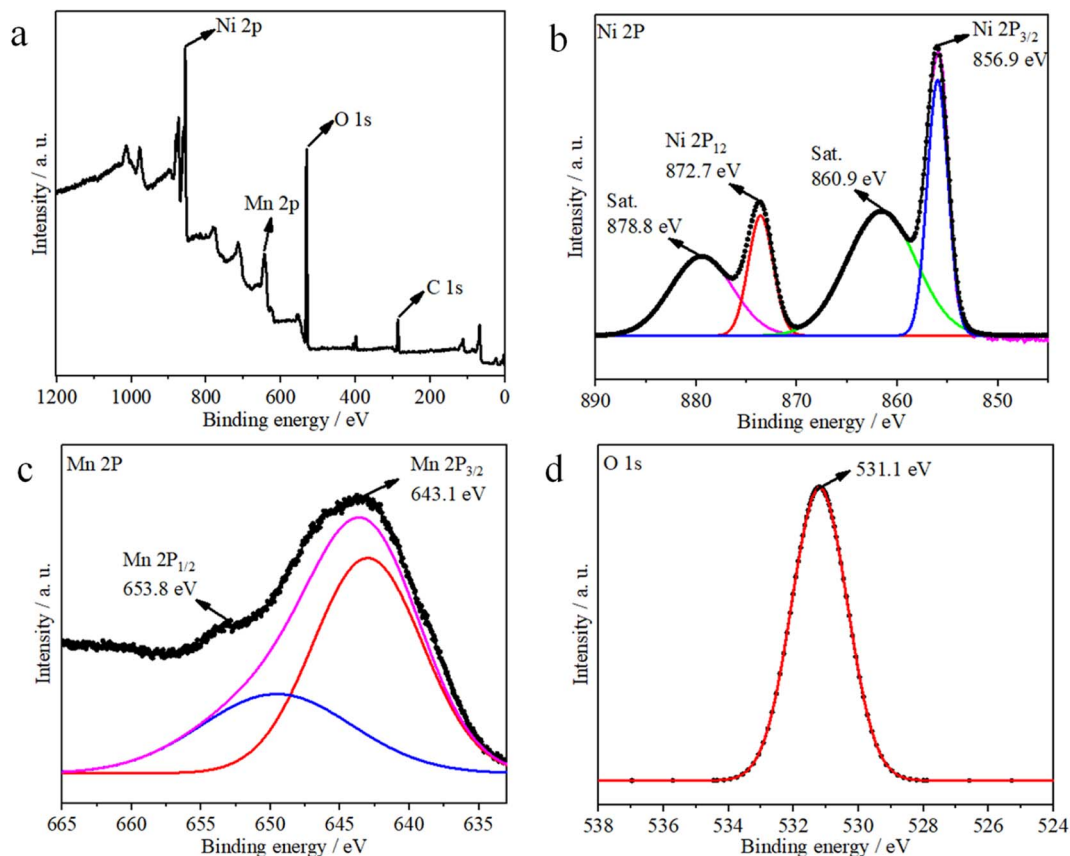


Fig. 3 XPS survey spectra of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$: (a) full spectra, (b) Ni 2p, (c) Mn 2p and (d) O 1s.



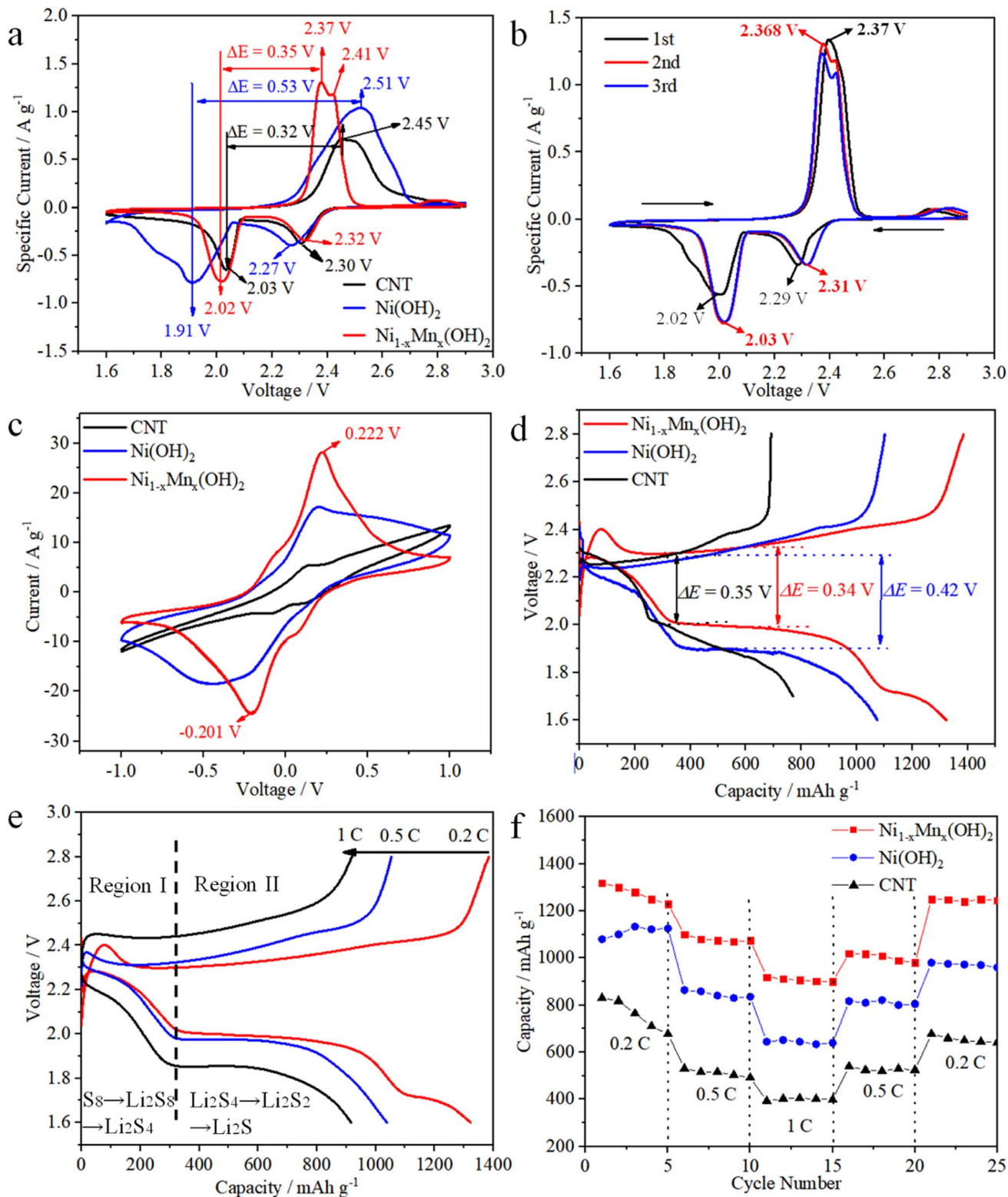


Fig. 4 CV curves of cells containing (a) the variously fabricated electrodes and (b) the Ni_{1-x}Mn_x(OH)₂ electrode at 0.1 mV s⁻¹. (c) CV profiles of symmetric cells with different electrodes at 0.1 mV s⁻¹. (d) First charge/discharge profiles of different electrodes at 0.2C. (e) Charge/discharge voltage curves containing Ni_{1-x}Mn_x(OH)₂ at different rates. (f) Rate capabilities of different electrodes between 0.2 and 1C.

electrode (0.35 V), which could be caused by the higher conductivity of CNT than that of Ni_{1-x}Mn_x(OH)₂. The specific current of the CNT electrode was significantly smaller than that

of the Ni(OH)₂ and Ni_{1-x}Mn_x(OH)₂ electrodes, implying that Ni(OH)₂ and Ni_{1-x}Mn_x(OH)₂ have excellent electrochemical activity.^{23,24}

The initial three CV cycles of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode at 0.1 mV s^{-1} were tested in order to estimate the electrochemical stability (Fig. 4b). The two reduction peaks of the first cycle were at 2.32 and 2.02 V, and the oxidation peak was located at 2.51 V. In comparison, the two reduction peaks of the second cycle were at 2.31 and 2.03 V, and the oxidation peak was located at 2.368 V, implying that the cell had formed a stable solid electrolyte interface (SEI) film and irreversible decomposition of the electrolyte occurred.²⁵ The superior overlap of the second and third cycles of the CV curve proves the stability and better reversibility of the cell.

CV curves of the symmetrical cell based on the $\text{Ni}(\text{OH})_2$ electrode exhibited one pair of wider redox peaks resulting from the repeated reduction reactions of Li_2S_8 to short-chain LiPSs, revealing the sluggish electrochemical kinetics of $\text{Ni}(\text{OH})_2$ (Fig. 4c).^{26–28} The CV of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode exhibited one pair of distinct reversible peaks at -0.201 and 0.222 V , which displayed considerably higher current densities and smaller polarisation compared to the CNT and $\text{Ni}(\text{OH})_2$ electrodes. Thus, $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ effectively enhanced the redox kinetics of polysulphides more than CNT and $\text{Ni}(\text{OH})_2$.²⁹ The representative discharge/charge curves between 1.7 and 2.8 V at 0.2C are shown in Fig. 4d to demonstrate the role of various electrodes in Li–S batteries. The first discharge/charge capacity of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode was 1375/1386 mA h g^{-1} , and the curve of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode exhibited smaller capacity loss and more steady voltage platforms than that of the $\text{Ni}(\text{OH})_2$ electrode. Furthermore, ΔE (the potential gap between the charge and discharge plateaus) for the cell with $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode (0.34 V) was smaller than that of the $\text{Ni}(\text{OH})_2$ electrode (0.42 V). The decrease in ΔE demonstrates that the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode relieved the redox polarisation in the cells. Compared with semiconducting $\text{Ni}(\text{OH})_2$, the introduction of Mn primarily contributed to the presence of half-metallic $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, leading to ion and electron transmission, which further reduced charge-transfer resistance and improved electrochemical performance.³⁰ ΔE for the cell with the CNT electrode (0.35 V) was similar to that of the cell with the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode (0.34 V). However, the curve of the CNT electrode exhibited a larger capacity loss and less steady voltage platforms than those of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode, thus demonstrating that $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ effectively enhanced the redox kinetics of polysulphides. The discharge/charge curves of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode at 0.2, 0.5 and 1C are shown in Fig. 4e. The curve also maintained a stable flat discharge platform even at 1C, and exhibited light distortion at high current density.

The rate performances of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, $\text{Ni}(\text{OH})_2$ and CNT electrodes are depicted in Fig. 4f. The first discharge specific capacities of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode at 0.2, 0.5 and 1C were 1320, 1100 and 920 mA h g^{-1} , and the specific capacity returned to 1250 mA h g^{-1} when turned back to 0.2C. Nevertheless, the initial discharge specific capacities of the $\text{Ni}(\text{OH})_2$ electrode at 0.2, 0.5 and 1C were 1080, 863 and 645 mA h g^{-1} , and the reversible specific capacity was only 980 mA h g^{-1} when turned back to 0.2C. Obviously, the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode can achieve higher reversible redox reactions than the $\text{Ni}(\text{OH})_2$

electrode. The CNT electrode exhibited worse electrochemical performance than those of the two abovementioned electrodes because of the lack of functional materials.

Fig. 5a compares the cycling performances of the CNT, $\text{Ni}(\text{OH})_2$ and $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrodes at 0.2C. The first discharge specific capacities of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, $\text{Ni}(\text{OH})_2$ and CNT electrodes were 1323, 1050 and 690 mA h g^{-1} , and the reversible specific capacities of the above electrodes were 813, 472 and 350 mA h g^{-1} , respectively, after 200 cycles. The $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode exhibited a more stable cycling performance than the $\text{Ni}(\text{OH})_2$ electrode; this performance was substantially higher than the CNT electrode. The initial discharge specific capacity of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode at 0.5C was 1150 mA h g^{-1} , which subsequently increased to 1283 mA h g^{-1} ; the reversible specific capacity was 714 mA h g^{-1} after 200 cycles. The discharge specific capacities slightly increased between the second and fifth cycle, probably because of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ undergoes an initial activation process during discharge/charge.^{8,31} In comparison, the discharge specific capacity of the $\text{Ni}(\text{OH})_2$ electrode decreased to 474 mA h g^{-1} after 200 cycles, which was still considerably higher than that of the CNT electrode. The decay trends of the discharge specific capacity of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ and CNT electrodes were similar under 0.2 and 0.5C because of the perfect conductivity and minor polarisation of the two abovementioned compounds. Conversely, the $\text{Ni}(\text{OH})_2$ electrode showed inferior cyclic stability because of the worse conductivity of $\text{Ni}(\text{OH})_2$.

EIS was performed on the cells after 200 cycles based on $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, $\text{Ni}(\text{OH})_2$, and CNT electrodes, and the corresponding Nyquist plots are shown in Fig. 5c. Apparently, the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode had a considerably smaller semicircle at high frequency than the $\text{Ni}(\text{OH})_2$ electrode, implying that the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode had lower interface charge-transfer resistance (R_{ct} , 10.1 Ω) than the $\text{Ni}(\text{OH})_2$ electrode (R_{ct} , 48.4 Ω). These results imply that the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode is favourable for promoting ionic mobility and electronic conductivity on the electrolyte–electrode interface.³² In addition, the R_{ct} (13.8 Ω) of the CNT electrode was substantially smaller than that of the $\text{Ni}(\text{OH})_2$ electrode, confirming that the electronic conductivity of the CNT electrode is significantly higher than that of the $\text{Ni}(\text{OH})_2$ electrode and similar to that of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode. The $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode was significantly better than $\text{Ni}(\text{OH})_2$ electrode in terms of electrochemical activity based on the slopes of the three electrodes in the low-frequency region.

The capacity of solid Li_2S deposition from Li_2S_8 was tested using the variously prepared electrodes and test methods presented in Section 2.4. The results (Fig. 5d–f) show that the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode had the strongest current peak and the highest nucleation capacity of Li_2S (175 mA h g^{-1}). The CNT and $\text{Ni}(\text{OH})_2$ electrodes exhibited current peaks of 102 mA h g^{-1} and 105 mA h g^{-1} , respectively. The results suggest that the presence of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode could reduce the Li_2S nucleation energy, strengthen the adsorption of LiPSs, and accelerate the rapid phase transformation of LiPSs to solid Li_2S . Thus, $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ promoted the liquid–solid nucleation



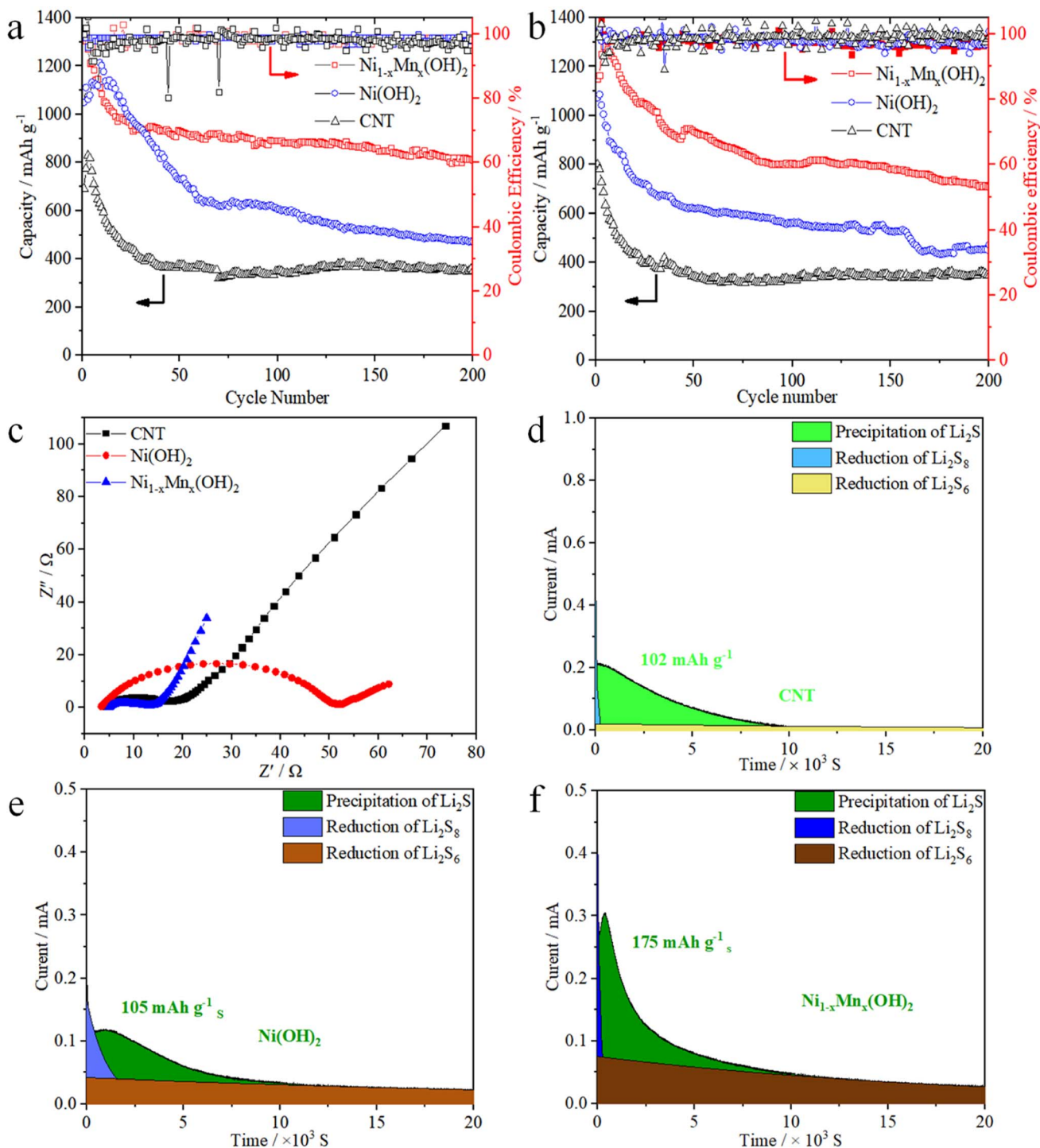


Fig. 5 Cycling properties of Li-S batteries containing Ni_{1-x}Mn_x(OH)₂, Ni(OH)₂ and CNT electrodes at (a) 0.2 and (b) 0.5C. (c) Nyquist plots of Li-S batteries containing Ni_{1-x}Mn_x(OH)₂, Ni(OH)₂ and CNT electrodes after 200 cycles. Potentiostatic discharge profiles of Li₂S deposition with cells containing (d) CNT, (e) Ni(OH)₂ and (f) Ni_{1-x}Mn_x(OH)₂ electrodes.

and growth kinetics of Li₂S.³³⁻³⁶ In a previous study, we discussed the key role of Ni(OH)₂ in improving the redox reaction kinetics in Li-S batteries.³⁷ However, Ni(OH)₂ also has some disadvantages, such as poor conductivity and few electrochemically active areas, which limit its kinetic diffusion. The O-coordinated Mn atoms doped in Ni(OH)₂ induce a large binding

energy between LiPSs and the active sites that accelerates redox reaction kinetics during the discharge/charge process.^{38,39} Similar to that in the CV test and EIS analysis, the relatively low Li₂S deposition on Ni(OH)₂ also resulted from the poorer conductivity of Ni(OH)₂ relative to Ni_{1-x}Mn_x(OH)₂.^{40,41}



The preparation of the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ cathode slurry was the same as those of the coin and pouch cells but with the electrodes of the pouch cell having a higher sulphur mass loading of 6.4 mg cm^{-2} . The first discharge capacity of the pouch cell was 105 mA h (Fig. 6b), and the corresponding specific capacity was 1150 mA h g^{-1} (matched with $4.37 \text{ mA h cm}^{-2}$, Fig. 6a). After 50 cycles, the pouch cell still maintained a markedly high areal capacity (about $3.12 \text{ mA h cm}^{-2}$). The folded pouch cell could still make 'DZU' LED light similar to the smooth pouch cell (Fig. 6c and d). Thus,

the pouch cell has certain application prospects in the fields of wearable electronic equipment and special equipment.

SEM images of the Li anode after cycles are shown in Fig. 6e–g to further investigate the effects of different electrodes with Li anodes. Compared with $\text{Ni}(\text{OH})_2$ and CNT, the surface of the Li anode cycled with $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ was smoother with only a few cracks, as shown in Fig. 6e, attributable to the faster redox kinetics of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ polysulphides. It also exhibited relatively stable cycling performance. Moreover, the surface of the Li anode with $\text{Ni}(\text{OH})_2$ was loose and mossy, as shown in Fig. 6f, caused by the slower redox kinetics of the $\text{Ni}(\text{OH})_2$, and

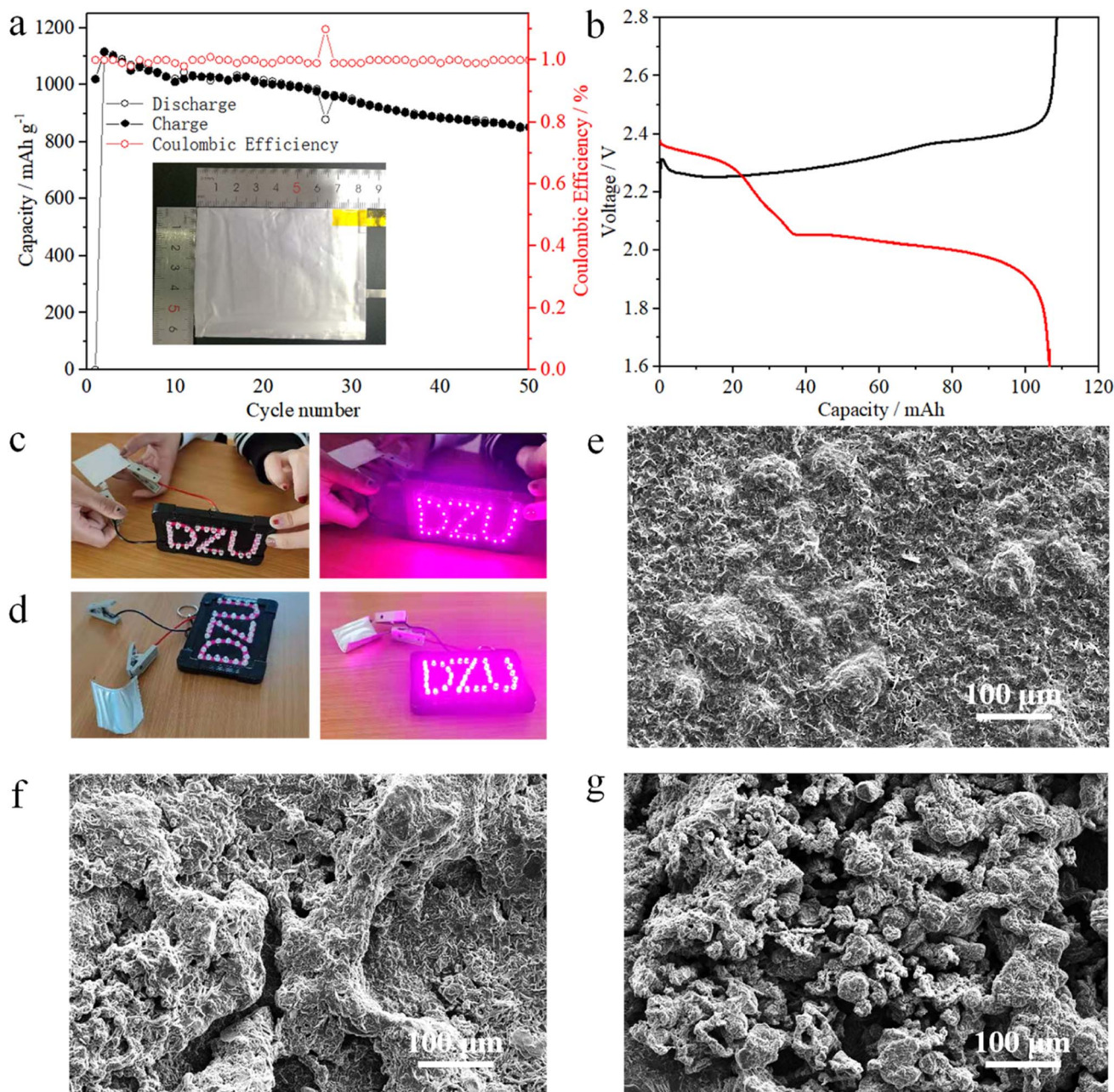


Fig. 6 Li–S pouch cell constructed with a $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode: (a) galvanostatic cycling performance and (b) voltage profile at 0.2C. The pouch cell with a $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode powers a pink LED logo under (c) flat and (d) folded states. SEM images of different Li metals after 100 cycles: (e) $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$, (f) $\text{Ni}(\text{OH})_2$ and (g) CNT.



the specific capacity decay was faster than that of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$. The surface of the Li anode with CNT showed numerous particles and even 'dead lithium' (Fig. 6g), resulting in worse electrochemical performance than that of $\text{Ni}(\text{OH})_2$.^{42,43} A comparison of the electrochemical performance of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ with that of carbon-based functional materials presented in previous literature is shown in Table S1:† $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ performed well among the electrodes that have been studied in the literature, evidencing the importance of Mn-doped $\text{Ni}(\text{OH})_2$ materials in improving electrochemical performance and enhancing S redox kinetics.

4. Conclusions

Petal-like $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ ($x \approx 0.04$) nanosheets were synthesised in this study *via* a simple hydrothermal method. With high S loading (about 5 mg cm^{-2}) on the cathode, the $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ electrode exhibited discharge specific capacities up to 1375 and 1150 mA h g^{-1} at 0.2 and 0.5C, respectively. The discharge specific capacity was retained at 813 and 714 mA h g^{-1} after 200 cycles, which exhibited better cycling stability and specific capacity in Li-S batteries. The results of the cyclic stability, symmetrical electrode and cells of CV measurements, EIS and the nucleation and growth of Li_2S prove that $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ plays a critical role in Li-S batteries compared with $\text{Ni}(\text{OH})_2$. The superior electrochemical performance of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ has two advantages. First, $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ has a larger specific surface area than $\text{Ni}(\text{OH})_2$, providing better adsorption performance for LiPSs. Second, the conductivity performance of $\text{Ni}_{1-x}\text{Mn}_x(\text{OH})_2$ was significantly improved compared with that of $\text{Ni}(\text{OH})_2$, resulting in the acceleration of the electrochemical reaction kinetics of the discharge/charge process.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the Science and Technology Development Fund of Shandong Province (2022TSGC2560) and Dezhou City (2022TSGC019).

References

- J. Q. Huang, X. F. Liu, Q. Zhang, C. M. Chen, M. Q. Zhao, S. M. Zhang, W. C. Zhu, W. Z. Qian and F. Wei, Entrapment of sulfur in hierarchical porous graphene for lithium-sulfur batteries with high rate performance from -40 to 60°C , *Nano Energy*, 2013, 2, 314–321.
- Y. C. Jeong, J. H. Kim, S. Nam, C. R. Park and S. J. Yang, Rational design of nanostructured functional interlayer/separator for advanced Li-S batteries, *Adv. Funct. Mater.*, 2018, 28, 1707411.
- S. Rehman, K. Khan, Y. F. Zhao and Y. L. Hou, Nanostructured cathode materials for lithium-sulfur batteries: progress, challenges and perspectives, *J. Mater. Chem. A*, 2017, 5, 3014–3038.
- W. G. Lim, S. Kim, C. Jo and J. Lee, A comprehensive review of materials with catalytic effects in Li-S batteries: enhanced redox kinetics, *Angew. Chem. Int. Ed.*, 2019, 58, 18746–18757.
- Y. Son, J. S. Lee, Y. Son, J. H. Jang and J. Cho, Recent advances in lithium sulfide cathode materials and their use in lithium sulfur batteries, *Adv. Energy Mater.*, 2015, 5, 1500110–1500123.
- J. P. Yue, M. Yan, Y. X. Yin and Y. G. Guo, Progress of the interface design in all-solid-state Li-S batteries, *Adv. Funct. Mater.*, 2018, 28, 1707533.
- H. Y. Shao, B. C. Huang, N. Q. Liu, W. K. Wang, H. Zhang, A. B. Wang, F. Wang and Y. Q. Huang, Modified separators coated with a $\text{Ca}(\text{OH})_2$ -carbon framework derived from crab shells for lithium-sulfur batteries, *J. Mater. Chem. A*, 2016, 4, 16627–16634.
- J. Jiang, J. Zhu, W. Ai, X. Wang, Y. Wang, C. Zou, W. Huang and T. Yu, Encapsulation of sulfur with thin-layered nickel-based hydroxides for long-cyclic lithium-sulfur cells, *Nat. Commun.*, 2015, 6, 8622–8630.
- C. L. Dai, L. Y. Hu, M. Q. Wang, Y. M. Chen, J. Han, J. Jiang, Y. Zhang, B. L. Shen, Y. B. Niu, S. J. Bao and M. W. Xu, Uniform $\alpha\text{-Ni}(\text{OH})_2$ hollow spheres constructed from ultrathin nanosheets as efficient polysulfide mediator for long-term lithium-sulfur batteries, *Energy Storage Mater.*, 2017, 8, 202–208.
- S. D. Min, C. J. Zhao, Z. M. Zhang, G. R. Chen, X. Z. Qian and Z. P. Guo, Synthesis of $\text{Ni}(\text{OH})_2/\text{RGO}$ pseudocomposite on nickel foam for supercapacitors with superior performance, *J. Mater. Chem. A*, 2015, 3, 3641–3650.
- J. Yan, Z. J. Fan, W. Sun, G. Q. Ning, T. Wei, Q. Zhang, R. F. Zhang, L. J. Zhi and F. Wei, Advanced asymmetric supercapacitors based on $\text{Ni}(\text{OH})_2/\text{graphene}$ and porous graphene electrodes with high energy density, *Adv. Funct. Mater.*, 2012, 22, 2632–2641.
- X. H. Sun, Q. Shao, Y. C. Pi, J. Guo and X. Q. Huang, A general approach to synthesise ultrathin NiM ($\text{M} = \text{Fe}, \text{Co}, \text{Mn}$) hydroxide nanosheets as high-performance low-cost electrocatalysts for overall water splitting, *J. Mater. Chem. A*, 2017, 5, 7769–7775.
- W. Y. Lim, Y. F. Lim and G. W. Ho, Pseudomorphic-phase transformation of NiCo based ternary hierarchical 2D-1D nanostructures for enhanced electrocatalysis, *J. Mater. Chem. A*, 2017, 5, 919–924.
- Y. Li, S. W. Guo, T. Jin, Y. L. Wang, F. Y. Cheng and L. F. Jiao, Promoted synergy in core-branch $\text{CoP}@/\text{NiFe-OH}$ nanohybrids for efficient electrochemical-photovoltage-driven overall water splitting, *Nano Energy*, 2019, 63, 103821.
- J. Y. Hwang, H. M. Kim, S. Shin and Y. K. Sun, Designing a high-performance lithium-sulfur batteries based on layered double hydroxides-carbon nanotubes composite cathode and a dual-functional graphene-polypropylene- Al_2O_3 separator, *Adv. Funct. Mater.*, 2018, 28, 1704294.
- G. Y. Jiang, N. Zheng, X. Chen, G. Y. Ding, Y. H. Li, F. G. Sun and Y. S. Li, In-situ decoration of MOF-derived carbon on nitrogen-doped ultrathin MXene nanosheets to



- multifunctionalize separators for stable Li-S batteries, *Chem. Eng. J.*, 2019, **373**, 1309–1318.
- 17 D. Guo, F. W. Ming, H. Su, Y. Q. Wu, W. D. Wahyudi, M. L. Li, M. N. Hedhili, G. Sheng, L. J. Li, H. N. Alshareef, Y. X. Li and Z. P. Lai, MXene based self-assembled cathode and antifouling separator for high-rate and dendrite-inhibited Li-S battery, *Nano Energy*, 2019, **61**, 478–485.
- 18 M. Yang, Y. Q. Wu, R. C. Rao and H. Wang, Methanol promoted synthesis of porous hierarchical a-Ni(OH)₂ for the remove of Congo red, *Powder Technol.*, 2017, **320**, 377–385.
- 19 J. W. Wu, W. Liu, X. Xiang, K. Sun, F. H. Liu, C. Cai, S. B. Han, Y. Y. Xie, S. Li and X. T. Zu, From Ni(OH)₂/Graphene composite to Ni@Graphene core-shell: a selfcatalyzed epitaxial growth and enhanced activity for nitrophenol reduction, *Carbon*, 2017, **117**, 192–200.
- 20 T. H. Xu, G. Y. Li, X. H. Yang, Z. X. Guo and L. J. Zhao, Design of the seamless integrated C@NiMn-OH-Ni₃S₂/Ni foam advanced electrode for supercapacitors, *Chem. Eng. J.*, 2019, **362**, 783–793.
- 21 Y. J. Wei, L. Y. Yan, C. Z. Wang, X. G. Xu, F. Wu and G. Chen, Effects of Ni doping on [MnO₆] octahedron in LiMn₂O₄, *J. Phys. Chem. B*, 2004, **108**, 18547.
- 22 J. W. Zhao, J. L. Chen, S. M. Xu, M. F. Shao, Q. Zhang, F. Wei, J. Ma, M. Wei, D. G. Evans and X. Duan, Hierarchical NiMn layered double hydroxide/carbon nanotubes architecture with superb energy density for flexible supercapacitors, *Adv. Funct. Mater.*, 2014, **24**, 2938–2946.
- 23 J. Cao, C. Chen, Q. Zhao, N. Zhang, Q. Lu, X. Wang, Z. Niu and J. Chen, A flexible nanostructured paper of a reduced graphene oxide-sulfur composite for high-performance lithium-sulfur batteries with unconventional configurations, *Adv. Mater.*, 2016, **28**, 9629–9636.
- 24 H. J. Peng, G. Zhang, X. Chen, Z. W. Zhang, W. T. Xu, J. Q. Huang and Q. Zhang, Enhanced electrochemical kinetics on conductive polar mediators for lithium-sulfur batteries, *Angew. Chem., Int. Ed.*, 2016, **128**, 13184–13189.
- 25 Y. Jiang, H. Q. Liu, X. H. Tan, L. M. Guo, J. T. Zhang, S. N. Liu, Y. J. Guo, J. Zhang, H. F. Wang and W. G. Chu, Monoclinic ZIF-8 nanosheet-derived 2D carbon nanosheets as sulfur immobilizer for high-performance lithium sulfur batteries, *ACS Appl. Mater. Interfaces*, 2017, **9**, 25239–25249.
- 26 Y. Li, X. Zhang, G. H. Liu, A. Gerhardt, K. Evans, A. Z. Jia and Z. S. Zhang, Oxygen-deficient titanium dioxide supported cobalt nano-dots as efficient cathode material for lithium-sulfur batteries, *J. Energy Chem.*, 2020, **48**, 390–397.
- 27 H. Y. Zhou, Z. Y. Sui, K. Amin, L. W. Lin, H. Y. Wang and B. H. Han, Investigating the Electrocatalysis of a Ti₃C₂/Carbon Hybrid in Polysulfide Conversion of Lithium-Sulfur Batteries, *ACS Appl. Mater. Interfaces*, 2020, **12**, 13904–13913.
- 28 F. Zhou, Z. Li, X. Luo, T. Wu, B. Jiang, L. L. Lu, H. B. Yao, M. Antonietti and S. H. Yu, Low Cost Metal Carbide Nanocrystals as Binding and Electrocatalytic Sites for High Performance Li-S Batteries, *Nano Lett.*, 2018, **18**, 1035–1043.
- 29 Z. Yuan, H. J. Peng, T. Z. Hou, J. Q. Huang, C. M. Chen, D. W. Wang, X. B. Cheng, F. Wei and Q. Zhang, Powering lithium-sulfur battery performance by propelling polysulfide redox at sulfiphilic hosts, *Nano Lett.*, 2016, **16**, 519–527.
- 30 J. W. Chen, X. Wang, J. X. Wang and P. S. Lee, Sulfidation of NiMn-Layered Double Hydroxides/Graphene Oxide Composites toward Supercapacitor Electrodes with Enhanced Performance, *Adv. Energy Mater.*, 2016, **6**, 1501745.
- 31 X. X. Gu, C. J. Tongc, B. Wen, L. M. Liu, C. Lai and S. Q. Zhang, Ball-milling synthesis of ZnO@sulphur/carbon nanotubes and Ni(OH)₂@sulphur/carbon nanotubes composites for high-performance lithium-sulphur batteries, *Electrochim. Acta*, 2016, **196**, 369–376.
- 32 Q. Li, Z. A. Zhang, Z. P. Guo, Y. Q. Lai and K. Zhang, Improved cyclability of lithium-sulfur battery cathode using encapsulated sulfur in hollow carbon nanofiber@nitrogen-doped porous carbon core-shell composite, *Carbon*, 2014, **78**, 1–9.
- 33 G. Xia, Z. Q. Zheng, J. J. Ye, X. T. Li, M. J. Biggs and C. Hu, Carbon microspheres with embedded FeP nanoparticles as a cathodeelectrocatalyst in Li-S batteries, *Chem. Eng. J.*, 2021, **406**, 126823.
- 34 H. C. Gao, S. L. Ning, J. S. Zou, S. Men, Y. Zhou, X. J. Wang and X. W. Kang, The electrocatalytic activity of BaTiO₃ nanoparticles towards polysulfides enables high-performance lithium-sulfur batteries, *J. Energy Chem.*, 2020, **48**, 208–216.
- 35 S. Z. Huang, X. M. Zhang, Y. Wang, Y. Zheng, D. Z. Kong, M. Ding, S. Y. Yang and H. Y. Yang, Regulating the polysulfide redox conversion by iron phosphide nanocrystals for high-rate and ultrastable lithium-sulfur battery, *Nano energy*, 2018, **51**, 340–348.
- 36 S. Fan, S. Z. Huang, M. E. Pam, S. Chen, Q. Y. Wu, J. P. Hu, Y. Wang, L. K. Ang, C. C. Yan, Y. M. Shi and H. Y. Yang, Design multifunctional catalytic interface: toward regulation of polysulfide and Li₂S redox conversion in Li-S batteries, *Small*, 2019, 1906132.
- 37 C. F. Zhao, X. X. Yang, G. Xia, J. L. Liu, W. Q. Zhang, J. Xue, S. F. Hou and C. Hu, Enhanced sulfur redox kinetics and polysulfide regulations with petal-like nickel hydroxide nanosheets/rGO modified separators in Li-S batteries, *Appl. Surf. Sci.*, 2021, **551**, 149393.
- 38 Y. N. Liu, Z. Y. Wei, B. Zhong, H. T. Wang, L. Xia, T. Zhang, X. M. Duan, D. C. Jia, Y. Zhou and X. X. Huang, O-, N- Coordinated single Mn atoms accelerating polysulfides transformation in lithium-sulfur batteries, *Energy Storage Mater.*, 2021, **35**, 12–18.
- 39 S. M. Qiao, D. Lei, Q. Wang, X. S. Shi, Q. Zhang, C. H. Huang, A. M. Liu, G. H. He and F. X. Zhang, Etch-evaporation enabled defect engineering to prepare high-loading Mn single atom catalyst for Li-S battery applications, *Chem. Eng. J.*, 2022, **442**, 136258.
- 40 Y. S. He, M. J. Li, Y. G. Zhang, Z. Z. Shan, Y. Zhao, J. D. Li, G. H. Liu, C. Y. Liang, Z. Bakenov and Q. Li, All-purpose electrode design of flexible conductive scaffold toward high-performance Li-S batteries, *Adv. Funct. Mater.*, 2020, **30**, 2000613.



Paper

- 41 M. Zhao, H. J. Peng, B. Q. Li, X. Chen, J. Xie, X. Y. Liu, Q. Zhang and J. Q. Huang, Electrochemical phase evolution of metal-based pre-catalysts for high-rate polysulfide conversion, *Angew. Chem., Int. Ed.*, 2020, **132**, 9096–9102.
- 42 C. Yan, X. B. Cheng, C. Z. Zhao, J. Q. Huang, S. T. Yang and Q. Zhang, Lithium metal protection through in-situ formed solid electrolyte interphase in lithium-sulfur batteries: The role of polysulfides on lithium anode, *J. Power Sources*, 2016, **327**, 212–220.
- 43 L. Qie, C. X. Zu and A. Manthiram, A high energy lithium-sulfur battery with ultrahigh-loading lithium polysulfide cathode and its failure mechanism, *Adv. Energy Mater.*, 2016, **6**, 1502459.

