


 Cite this: *RSC Adv.*, 2023, **13**, 18934

Appealing perspectives of the structural, electronic, elastic and optical properties of LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites: a DFT study

 Nasir Rahman,^{id}*^a Mudasser Husain,^{*b} Vineeth Tirth,^{id}^{cd} Ali Algahtani,^{cd} Hassan Alqahtani,^e Tawfiq Al-Mughanam,^f Abdulaziz H. Alghtani,^g Rajwali Khan,^a Mohammad Sohail,^a Abid Ali Khan,^h Ahmed Azzouz-Rached^{id}ⁱ and Aurangzeb Khan^j

To enhance the effectiveness of materials, we are motivated to investigate lithium-based halide perovskites LiRCl_3 (where $R = \text{Be}$ and Mg) using first-principles techniques based on density functional theory (DFT), implemented in the WIEN2K code. In this study, the research makes use of the WIEN2K simulation code, employing the plane-wave and self-consistent (PWSCF) approach. The cut-off energy, responsible for distinguishing core and valence states, is established at -6.0 Ry. To guarantee well-converged solutions with 2000 K points, parameters of $\text{RMT} \times K_{\text{max}} = 7.0$ are selected, where RMT represents the smallest muffin-tin radius and K_{max} denotes the plane wave cut-off. Convergence is determined to be attained when the overall energy of the system remains unchanged during self-consistent calculations, reaching a threshold of 0.001 Ry. We observe structural stability of these materials using the Birch–Murnaghan fit, tolerance factor and formation energy. The tolerance factor for LiMgCl_3 and LiBeCl_3 are 1.03 and 0.857, while the formation energy for LiMgCl_3 and LiBeCl_3 are -7.39 eV and -8.92 eV respectively, confirming these to be stable structurally. We evaluate the electronic properties of the current materials, shedding light on their nature, by using the suggested modified Becke–Johnson potential. It turns out that they are indirect insulators, with calculated band gaps of 4.02 and 4.07 eV for LiMgCl_3 and LiBeCl_3 , respectively. For both materials, we also calculate the density of states (DOS), and our findings regarding the band gap energies are consistent with the band structure. It is observed that both materials exhibit transparency to low-energy photons, with absorption and optical conduction occurring in the UV range. These compounds are mechanically stable, according to the elastic investigation, however LiBeCl_3 shows higher resistance to compressive and shear loads as well as resistance to shape change. On the other hand, LiMgCl_3 exhibits weaker resistance to changes in volume. Furthermore, we discovered that none of the compounds are entirely isotropic, and specifically, LiMgCl_3 and LiBeCl_3 are brittle in nature. These materials appear to be potential candidates for use in optoelectronic devices based on our analysis of their optical properties. Our findings may provide comprehensive insight, invoking experimental studies for further investigations.

 Received 21st April 2023
 Accepted 14th June 2023

DOI: 10.1039/d3ra02640j

rsc.li/rsc-advances

1. Introduction

Conducting theoretical studies on technologically significant materials offers a convenient approach to investigate their

properties prior to laboratory synthesis. Materials scientists are continuously seeking to discover the best and most efficient materials, even in a society that is already well-equipped with scientific knowledge. Due to their significant uses in the

^aDepartment of Physics, University of Lakki Marwat, 28420, Lakki Marwat, KPK, Pakistan. E-mail: nasir@ulm.edu.pk

^bState Key Laboratory of Mesoscopic Physics and Department of Physics, Peking University, 100871, Beijing, People's Republic of China. E-mail: 2201110247@stu.pku.edu.cn

^cMechanical Engineering Department, College of Engineering, King Khalid University, Abha 61421, Asir, Kingdom of Saudi Arabia

^dResearch Center for Advanced Materials Science (RCAMS), King Khalid University, Guraiger, Abha-61413, Asir, Kingdom of Saudi Arabia

^eDepartment of Mechanical Engineering, Taibah University, Medina 42353, Kingdom of Saudi Arabia

^fDepartment of Mechanical Engineering, College of Engineering, King Faisal University, P. O. Box 380, Al-Ahsa 31982, Saudi Arabia

^gDepartment of Mechanical Engineering, College of Engineering, Taif University, P.O. Box 11099, Taif 21944, Kingdom of Saudi Arabia

^hDepartment of Chemical Sciences, University of Lakki Marwat, 28420, Lakki Marwat, KPK, Pakistan

ⁱMagnetic Materials Laboratory, Faculty of Exact Sciences, Djillali Liabes University of Sidi Bel-Abbes, Algeria

^jDepartment of Physics, Abdul Wali Khan University, Mardan, 23200, KPK, Pakistan



semiconductor, energy storage, and optical industries, a lot of interest has been shown in perovskites.^{1–7} Halide perovskites fall within a sub-category of perovskites characterized by the general formula ABX_3 , where A represents a monovalent cation, B denotes a metal, and X represents an anion from the halogen group.⁸ Extensive investigations have been conducted on halide perovskites with diverse chemical compositions for various applications. Their non-birefringent nature, which eliminates difficulties in lens design, makes perovskite materials a preferred choice for lenses.⁹ In the chemical formula (ABX_3) of halide perovskite, the cations X and B represent alkali and transition metals, respectively. Perovskite compounds have an extensive variety of uses because of their distinct characteristics, which comprise a tunable laser,¹⁰ electron–phonon interactions,^{11,12} ferroelectricity,¹³ crystal fields,¹⁴ phase transition behaviors,¹⁵ and other physical properties like anti-ferromagnetism¹⁶ and semi-conductivity,¹⁷ and as scintillating materials.¹⁸ Scintillating materials and scintillators find widespread use in various fields such as healthcare (medical imaging),¹⁹ homeland security,²⁰ prototype formation,²¹ and astronomy,²² *etc.*, possessing a great level of intensity, directionality, and energy.²³ Scintillators have ability to transform ionizing X-rays or photons into light that is visible or nearly visible (near-ultraviolet). The electrical signal that is created from the light emission can subsequently be analyzed based on the applications. Recently, chloroperovskites has gained recognition as a scintillating substance due to their appropriate bandgaps, high absorption coefficients, and affordable manufacture.^{24–26} In recent developments, the inclusion of alkali cations such as Li, Na, K, Rb, and Cs in perovskite solar cells has led to improved photovoltaic performance, resulting in high power conversion efficiency.^{27–29} The first-principles study on alkaline metals (Li, K, Na, and Rb) based on lead halide perovskite structures was carried out by Atsushi Suzuki and Takeo Oku.³⁰ A notable influence of these alkali metal variations on electronic and thermodynamic properties was observed by them. The introduction of alkali metals into the perovskite crystal resulted in improved photo-induced carrier generation and diffusion within the perovskite layer's crystalline domain, leading to enhanced long-term stability under ambient conditions.³¹ The latest investigations into halide perovskites incorporating alkali metal ions have recognized their potential in optimizing structural and optoelectronic properties. This includes maximizing luminance, facilitating charge injection, improving moisture resistance, and enhancing energy band alignment, particularly in polycrystalline and nano-crystalline thin films.^{32–34} Nonetheless, extensive theoretical investigations have been undertaken by researchers to establish the presence and importance of semiconducting materials. Based on initial first-principles calculations, researchers have successfully showcased the confinement of structures through the utilization of metal organic chemical vapor deposition (MOCVD). This deposition technique is widely employed in the fabrication of semiconductor materials and has been instrumental in validating the confinement of these structures.^{35–38} Similar techniques can indeed be employed to synthesize various chloroperovskites compounds, enabling the fabrication of high-

performance semiconducting devices. These compounds, which typically consist of a perovskite structure with chlorine atoms incorporated into the lattice, have attracted significant attention in the field of optoelectronics. Methods such as solution processing, vapor deposition, and solid-state reactions can be utilized to synthesize chloroperovskites materials with precise control over composition and morphology. The resulting compounds can then be integrated into devices such as solar cells, light-emitting diodes (LEDs), and photo-detectors, showcasing their potential for achieving enhanced device performance. Although there has been remarkable progress and tremendous potential in this emerging technology, the commercialization of halide perovskites still encounters significant challenges. These challenges arise from the inclusion of toxic “Pb” and the material's susceptibility to instability against moisture, heat, and irradiation.³⁹ The structural and mechanical stability of perovskite materials plays a crucial role in their large-scale production for commercial purposes. Researchers are actively investigating and exploring the performance of these materials in various environments. The focus is primarily on the optoelectronic characteristics relevant to photovoltaic applications. Density Functional Theory (DFT) serves as a valuable tool for studying different properties of perovskite materials. The reliable and consistent results obtained from DFT-based computational studies demonstrate the credibility of this approach.⁴⁰ Various techniques can be employed to modify the characteristics of perovskite materials for specific applications. External electric and magnetic fields, stress, heat treatment, doping, and variation in constituents and vacancies have a notable impact on the physical properties of these materials. Nevertheless, there is currently no existing theoretical or experimental research available on the cubic phase of $LiBeCl_3$ and $LiMgCl_3$ compounds in the literature. A comprehensive literature review on halide perovskites suggests that the elements “Be” and “Mg” likely play a significant role in determining the structural, elastic, and optoelectronic properties of the ternary $LiRCl_3$ ($R = Be$ and Mg) halide perovskites. Using the DFT framework, this study is designed to present novel insights into the fundamental physical characteristics of ternary $LiRCl_3$ (where $R = Be$ and Mg) halide perovskites, with the potential application in energy storage devices and scintillating materials, and a variety of contemporary technological gadgets.

2. Computational methodology

The presented research employs the plane-wave and self-consistent (PWSCF) approach, based on DFT,⁴¹ using the simulation code WIEN2K.⁴² To compute the elastic characteristics and basic structural properties for crystal structure optimization, (GGA-PBE)⁴³ is utilized. For the computation of various elastic property parameters, the IRelast package⁴⁴ is employed. To enhance the accuracy of electronic properties and optical properties, the study utilizes the (TB-mBJ) for exchange–correlation potential calculations.⁴⁵ The Tb-mBJ exchange correlation potential is a modified form of the exchange–correlation potential used in density functional theory (DFT)



calculations. It is an approximation that combines the Tran–Blaha modified Becke–Johnson (mBJ) exchange potential with the standard DFT exchange–correlation potential. In reciprocal space, the primary Brillouin zone denotes sampled using 2000 K -points. For achieving a decent degree of convergence, this study examines several The FP-LAPW basis functions are utilized up to the product of RMT and K_{\max} equal to 7. The size of the greatest k -vector in the plane wave expansions is determined by K_{\max} . Each atom is associated with its own distinct RMT value to avoid any overlap. For computing the optical characteristics, the study employs the basic complex dielectric function in the form of $\epsilon(\omega) = \epsilon_1(\omega) + i\epsilon_2(\omega)$.⁴⁶ The primary graphing tools utilised in research are xcrysden,⁴⁷ Gnuplot,⁴⁸ origin⁴⁹ and xmgrace.⁵⁰ The computational method used in this work reveals, the physical properties are being investigated for the first time of ternary LiRCl₃ (R = Be and Mg) halide perovskites, including their structural, elastic, optical, and electrical characteristics, are revealed by the use of the computational method indicated above. In this particular study, the WIEN2K simulation code is employed, utilizing the plane-wave and self-consistent (PWSCF) approach. The cut-off energy, utilized to differentiate between core and valence states, is set at -6.0 Ry. To ensure well-converged solutions with 2000 K points, the parameters $\text{RMT} \times K_{\max} = 7.0$ are chosen, where RMT represents the smallest muffin-tin radius and K_{\max} corresponds to the plane wave cut-off. Convergence is deemed achieved when the overall energy of the system remains constant throughout the self-consistent calculations, reaching a threshold of 0.001 Ry. To compute the total energy, a Monkhorst–Pack⁵¹ $6 \times 6 \times 6$ k -mesh is utilized for Brillouin zone integration. For the computation of the density of states (DOS) and optical properties, the k -mesh is increased to an $11 \times 11 \times 11$ configuration.

3. Results and discussion

Using the methods outlined above, the results for ternary LiRCl₃ (R = Be and Mg) halide perovskites are described and shown in detail in this section. Here, each physical characteristic is treated separately.

3.1. The structural and phonon properties

An important step in the *ab initio* simulation study aims to examine the structural properties of these materials, which can provide insights into their additional physical properties, for instance, electrical, optical, and elastic properties. In this section, we investigate the structural properties of LiRCl₃ (R = Be and Mg) halide perovskites that fall under the space group-221 are comprehended utilizing the GGA-PBE approximation ($Pm\bar{3}m$). Fig. 1 demonstrates the LiRCl₃ halide perovskites (R = Be and Mg) primitive crystallographic unit cell in which 5 atoms make up each crystalline unit cell, with positions of Li, Be/Mg are (0, 0, 0), (0.5, 0.5, 0.5), and Cl is located at (0, 0.5, 0.5) respectively. By utilizing the Birch–Murnaghan equation of state, this study obtains key structural parameters, including bulk modulus, optimized lattice constant, pressure derivative of bulk modulus, unit cell optimum volume, and unit cell

optimum energy, through fitting the energy of the primitive unit cell as a function of its volume parameters,⁵² as depicted below:

$$E(V) = E_0 + \frac{B}{B'(B'-1)} \left[V \left(\frac{V_0}{V} \right)^{B'} - V_0 \right] + \frac{B}{B'} (V - V_0) \quad (1)$$

where in the equation above E_0 , B , B' , V , and V_0 are respectively, the optimum value of total energy, the bulk modulus, the pressure derivative of bulk modulus, the total unit cell volume, and the optimized unit cell volume. The eqn (1) presented above represents the optimal values of several key structural properties for LiRCl₃ (R = Be and Mg) halide perovskites. Specifically, the variables E_0 , B , B' , V , and V_0 correspond to the structural parameters as described above. These parameters are crucial for studying material behavior in electronics, optics, and elasticity.

In Fig. 2 shows the optimization curves for structural analysis obtained through the GGA approximation have been computed, and Table 1 lists the optimized values obtained from the curves fitted with the Birch–Murnaghan method. The graph illustrates that as the volume increases up to a specific value (V_0), there is a corresponding decrease in the energy of the unit cell. Subsequently, as the volume further increases, the energy of the system begins to rise, indicating an unstable state. In first principle computations, the structural parameters are determined based on the most stable state of the unit cell, which corresponds to the minimum energy state.

The following expression is used to compute the energies of formation for the ternary compound LiRCl₃ (R = Be and Mg) halide perovskites in the cubic phase, which demonstrated the chemical stability of the intricate compounds.

$$\Delta H_f = E_{\text{total}}(\text{LiRCl}_3) - E_{\text{Li}} - E_{\text{R}} - 3E_{\text{Cl}} \quad (2)$$

where in the aforementioned eqn (2), H_f is the energy of formation, $E_{\text{Total}}(\text{LiRCl}_3)$ is the optimal total energy of LiRCl₃ (R = Be and Mg), E_{R} is the energy of the “Be and Mg” atom’s ground state, E_{Li} represents the lowest energy state of the “Li” atoms, while E_{Cl} represents the lowest energy state of the chlorine atom. Table 1 presents the estimated values for the formation energies of interested compounds. All of the results are negative, demonstrating that the chloroperovskite compounds LiRCl₃ (R = Be and Mg) are stable structurally.^{2,53,54}

3.1.1. Phonon properties. The phonon dispersion curves provide information about the vibrational modes and properties of a material. The phonon band structure provides valuable insights into the thermal and mechanical properties of materials, including heat conduction, thermal expansion, and lattice stability. It is an essential tool for understanding and designing materials for various applications in solid-state physics, materials science, and engineering. In a phonon dispersion curve, different branches or bands can be observed, each corresponding to different vibrational modes of the crystal lattice. The acoustic branches represent the low-energy, long-wavelength vibrations that are associated with the collective motion of the lattice. In the phonon band structure plot, the horizontal axis typically represents the wave vector or



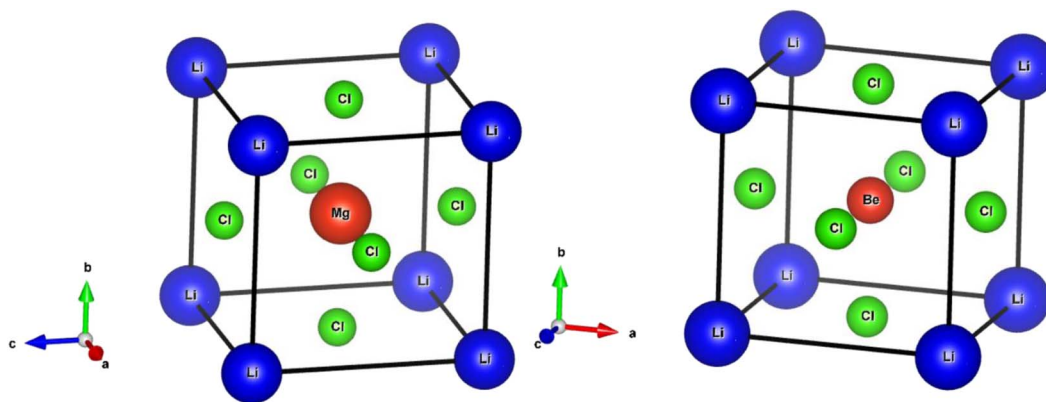


Fig. 1 The investigated crystal structures of ternary LiRCl_3 ($R = \text{Be}$ and Mg) chloroperovskites compounds.

momentum, while the vertical axis represents the phonon energy. The plot consists of multiple branches or bands, each corresponding to different vibrational modes. The calculated phonon spectra are depicted in Fig. 3, and it is very obvious that the positive mode of vibration occurs for both the materials. Positive frequencies in a phonon dispersion curve indicate physically allowed and meaningful vibrational modes of the crystal lattice. They provide information about the energy and behavior of phonons and are essential for understanding the lattice dynamics and properties of materials.

3.2. The electronic properties

The electronic band structure and DOS of chloroperovskite materials determine their electronic properties, such as whether they exhibit metallic, semiconducting, or insulating behavior, as well as the types of bonding that occur between their constituent elements and their corresponding gap energies. The band structure provides insights into material conductivity and acts as a crucial link between crystal structure and electronic/optical properties. Studying electronic properties is crucial for a comprehensive understanding of these materials. DFT computations can be used to determine a material's band structure after optimising its structure using 2000 K -

Table 1 The investigated optimized structural parameters for ternary LiRCl_3 ($R = \text{Be}$ and Mg) chloroperovskites exhibit various physical properties, such as lattice constant a_0 (in Å), bulk modulus B (in GPa), pressure derivative of bulk modulus B' , optimal volume V_0 (in atomic units)³, ground state energy E_0 (in Ry), and formation energies H_f (in eV)

Optimized structural parameters	LiMgCl_3	LiBeCl_3
a_0 in Å	4.95	4.56
E_0 in Ry	-3185.298454	-2814.001590
V_0 in (a.u.) ³	821.0834	644.0256
B in GPa	32.79	44.60
B' in GPa	4.2723	4.20
τ	1.03	0.857
ΔH_f (eV)	-7.39	-8.92

points in the first Brillouin Zone. Both of the studied ternary LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites compounds have band structures that are anticipated by the TB-mBJ approximation to be in the Brillouin zone between -14 eV and +8 eV, according to the high symmetry sites " $M-\Gamma$ " and " $T-\Gamma$ " illustrated in Fig. 4. The analysis of the band structure for both compounds indicates that the VB maxima are situated at the R symmetry point, whereas the CB minima are situated at the " T "

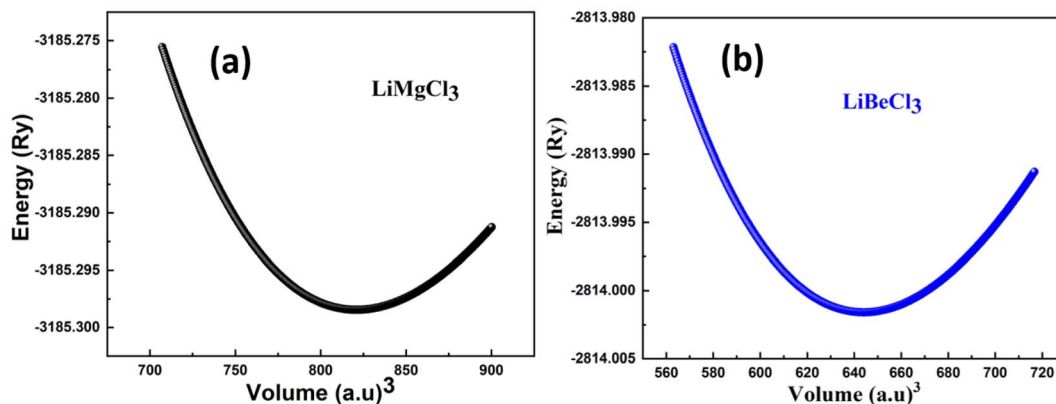


Fig. 2 The cubic-phase optimization of the energy of the primitive unit cell versus its volume LiRCl_3 ($R = \text{Be}$ and Mg) chloroperovskites.



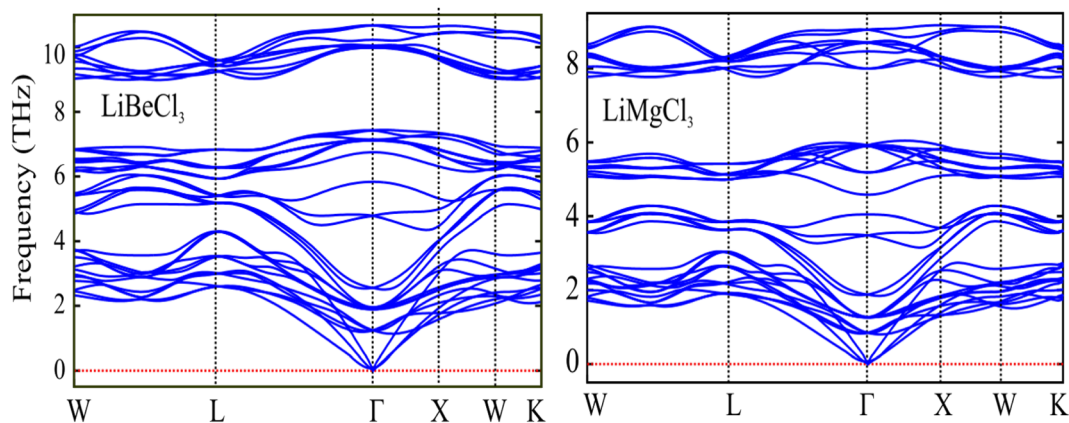


Fig. 3 The computed phonon band spectra of LiBeCl₃ and LiMgCl₃ ternary chloroperovskites compounds.

symmetry point. The calculated band gaps for LiMgCl₃ and LiBeCl₃ are 4.02 eV and 4.07 eV, respectively, indicating that both compounds are indirect band gap insulators. This suggests that these materials can behave as semiconductors when doped with heavy n-type materials, making them suitable for applications in semiconducting industries and technologies, such as energy storage devices and modern electronic gadgets.

To further explore the electronic properties, the analysis of DOS, is necessary to gain a thorough understanding of a material's electronic properties. This analysis is shown in Fig. 4 for LiRCl₃ (R = Mg and Be) halide perovskites in the energy range of

−14 eV to 8 eV. The Fermi energy (E_F) level is represented by the horizontal red dotted line at 0 eV in both graphs. In the electronic band structure portion above the Fermi level is referred to as the conduction band (CB), while the portion below the Fermi level is known as the valence band (VB). The density of states (DOS) provides a detailed analysis of the contribution of individual electronic states of the constituent elements to the VB and CB. The contributions from each Li, Mg and Cl is shown in Fig. 4(a), it is evident that in the conduction band the dominant contribution comes from Li, Mg and Cl, however, in the valence band the contribution comes from “Mg” element. The electronic band structure gap for LiRCl₃ (R = Mg and Be)

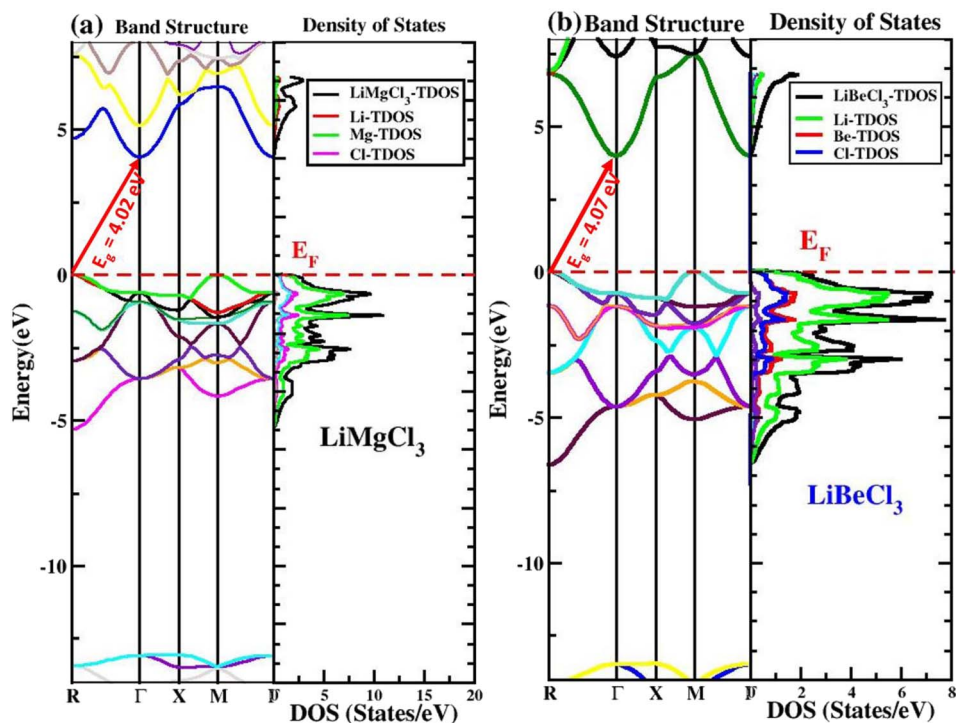


Fig. 4 The electronic band structures and density of states (DOS) for LiRCl₃ (R = Mg and Be) halide perovskites examined by fitting the data to high symmetry points in the first Brillouin zone, using the TB-mBJ approximation.



chloroperovskites is totally consistent with the band gap shown in DOS, thus confirming indirect insulating nature for both the materials.

3.2.1. Effective masses. Effective mass is an important parameter in determining the electrical and optical properties of materials, including conductivity, mobility, and energy band structures. The equation provided allows for the determination of the effective masses of the carriers.

$$m^* = \hbar^2 \left(\frac{\partial^2 E}{\partial K^2} \right)^{-1} \quad (3)$$

The simplified Planck constant, denoted as \hbar , and the wave vector represented by K , are utilized in equation⁵⁵ to calculate the eigenvalues of the energy band, denoted as E . Table presents the relative values of the hole effective masses (m_h^*) and electron effective masses (m_e^*) of LiBeCl₃ and LiMgCl₃ with respect to the mass of a free electron (m_0).

The small difference between the values of 1 minus m_e^*/m_h^* indicates a significantly high rate of recombination of electron-hole pairs. There exists an inverse relationship between the effective mass values of holes and electrons and the mobility of carriers. As the effective mass values increase, the mobility of carriers decreases. We observe a negative effective mass of holes, indicating the carriers exhibit unusual behavior. For example, they can accelerate in the opposite direction of an applied force or exhibit behaviors that seem counterintuitive. This phenomenon is typically associated with specific energy bands or band structures in materials. On the other hand, the effective mass of electrons in LiMgCl₃ is greater than that in LiBeCl₃, indicating that electrons in LiMgCl₃ encounter more resistance to movement or possess greater inertia in comparison to electrons in LiBeCl₃. In simpler terms, electrons in LiMgCl₃ are less mobile or more sluggish compared to those in LiBeCl₃.

3.3. The optical properties

The various parameters relating to the optical behaviour of LiRCl₃ chloroperovskites (R = Be and Mg) are described in great detail in this section. The electronic properties of LiBeCl₃ and LiMgCl₃ suggest that these compounds have an insulating nature with a wide indirect band gap. This makes them suitable for optical applications. To determine various optical properties such as reflectivity, refractive index, extinction coefficient, and absorption coefficient, the following formulas are used.⁵⁶

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega) \quad (4)$$

$$n(\omega) = \left[\frac{\varepsilon_1(\omega)}{2} + \frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}}{2} \right]^{\frac{1}{2}} \quad (5)$$

$$k(\omega) = \left[\frac{-\varepsilon_1(\omega)}{2} + \frac{\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}}{2} \right]^{\frac{1}{2}} \quad (6)$$

$$I(\omega) = \frac{2\omega}{c} k(\omega) \quad (7)$$

$$R(\omega) = \frac{(1-n)^2 + k^2}{(1+n)^2 + k^2} \quad (8)$$

$$\sigma(\omega) = \frac{2W \text{ eV } \hbar(\omega)}{E_0} \quad (9)$$

Using the optimized lattice constant and the TB-mBJ technique, all the fundamental optical properties are calculated from the dielectric function for incident photon energies ranging from 0 eV to 40 eV.

3.3.1. The real “ $\varepsilon_1(\omega)$ ” and imaginary “ $\varepsilon_2(\omega)$ ” part of the dielectric function. The determination of dielectric function is instructive for several optical aspects of LiBeCl₃ and LiMgCl₃. The complex dielectric function is a fundamental quantity that describes the response of a material to electromagnetic fields, and it includes both electronic and lattice contributions. It is used to study the interaction between photons and electrons in a material, as well as to describe how electromagnetic waves propagate in the material. The dielectric function is a crucial quantity in determining the optical properties of a material, as it provides information about the material's response to incident light at all photon energies. The real part of the dielectric functions makes it amply evident that the patterns in the spectra of the two materials are consistent with one another. The “ $\varepsilon_1(\omega)$ ” curves extend from 0 eV incident photon energy up to 40 eV, demonstrating that the curve approaches to a maximum positive peak of 5.9 for LiMgCl₃ and 6.8 for LiBeCl₃ at the low range of photon energy. At around 10 eV to 20 eV input photon energy, both LiRCl₃ (R = Be and Mg) halide perovskites reach negative maximum values of about -1 for LiBeCl₃ and -0.5 for LiMgCl₃, indicating 100% reflection in these energy ranges. Real static “ $\varepsilon_1(0)$ ” dielectric function value for LiMgCl₃ and LiBeCl₃ is 3. Understanding the imaginary part, “ $\varepsilon_2(\omega)$ ”, of the dielectric function is crucial for analyzing optical transitions from the valence band to the conduction band. In Fig. 5, the “ $\varepsilon_2(\omega)$ ” spectra for both studied materials are shown on the right side. The maximum peaks for LiMgCl₃ and LiBeCl₃ in the photon energy range of 0 eV to 42 eV are 6 and 7.5, respectively. The “ $\varepsilon_2(\omega)$ ” starts from zero at about 5.0 eV, which is an absorption edge, and reaches its highest value of 6 at 10 eV for LiMgCl₃ and 7.5 at 13 eV for LiBeCl₃, respectively. This indicates that absorption is maximum in the ultraviolet (UV)-visible (Vis) range. These materials show great promise for use in optoelectronic devices such as light-emitting diodes (LEDs) due to their UV-Vis absorption.

3.3.2. The optical conductivity. The optical conductivity of materials describes the relationship between the applied electric field and induced current density at a specific frequency. Fig. 6 displays the calculated optical conductivity spectrum for LiRCl₃ chloroperovskites compounds (R = Be and Mg) across incident photon energies from 0 eV to 40 eV. The figure illustrates that the optical conductivity emerges at photon energies corresponding to the electronic band gap energy. LiMgCl₃



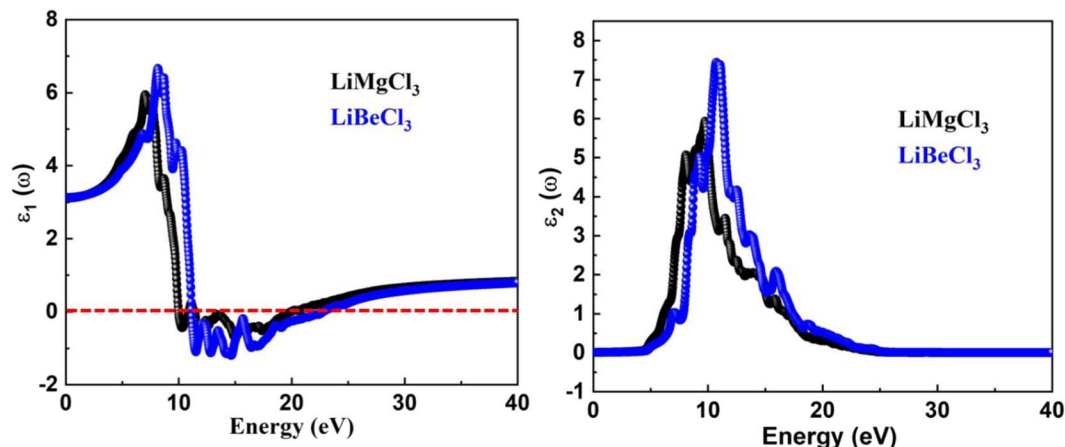


Fig. 5 The computed real and imaginary parts of dielectric function for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.

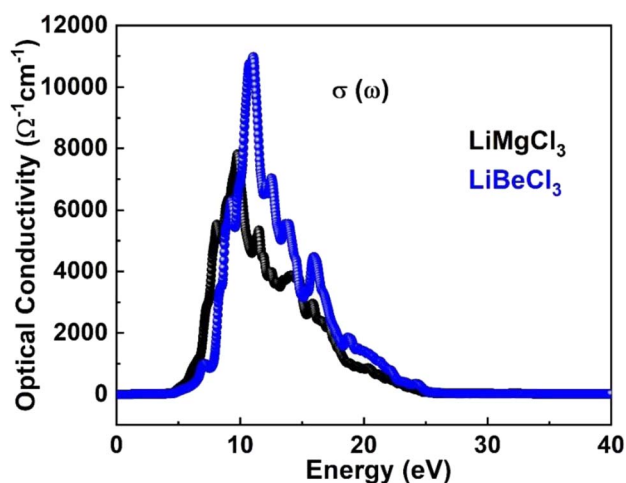


Fig. 6 The computed optical conductivity for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.

exhibits a maximum optical conductivity of $8000 \text{ Ohm}^{-1} \text{ cm}^{-1}$ at 10 eV photon energy, while LiBeCl_3 shows a maximum of $11000 \text{ Ohm}^{-1} \text{ cm}^{-1}$ at 13 eV photon energy. Both LiRCl_3 compounds demonstrate favorable optical conductivity at lower energies, suggesting their potential use in UV-Vis optical devices. Our results are consistent with the previous literature.^{2,5-7,57-66}

3.3.3. The absorption co-efficient. Eqn (6) quantifies the material's absorption per unit length of light,⁶⁷ the absorption coefficient spectrum $I(\omega)$ is determined using the TB-mBJ approximation and is displayed in Fig. 7. This illustrates the similarity between the behaviors of $I(\omega)$ and $\epsilon(\omega)$. Since electronic band transitions require energy above a certain threshold, it is expected that light absorption will not occur below this threshold.⁶⁸ The calculated band gap values for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites are smaller than the critical values of $I(\omega)$. At incident photon energies of 10 to 15 eV, the maximum absorption peaks are 160000 cm^{-1} for LiMgCl_3

and 220000 cm^{-1} for LiBeCl_3 . This indicates that these materials are suitable for applications in the UV-Vis range.

3.3.4. The extinction co-efficient. In Fig. 8, the extinction coefficient $K(\omega)$ spectrum for the investigated LiRCl_3 ($R = \text{Be}$ and Mg) chloroperovskite materials is shown across incident photon energies from 0 eV to 40 eV. For photon energies ranging from 0 to 5.0 eV, the extinction coefficient is zero for both compounds. As the photon energy increases, the extinction coefficient rises, followed by a significant decrease in the high-energy photon region. The extinction coefficient $K(\omega)$ quantifies the light loss per unit volume caused by scattering and absorption. The results indicate that these materials do not exhibit transparency to incident photons with lower energies.

3.3.5. The refractive index. Fig. 9 illustrates the refractive indices of LiMgCl_3 (black) and LiBeCl_3 (blue) within the energy range of 0–14 eV. Both materials have a static refractive index of 1.75. The curves for both materials coincide initially, but as the incident photon energy increases, the refractive indices increase as well. LiMgCl_3 reaches a maximum value of 2.5 at

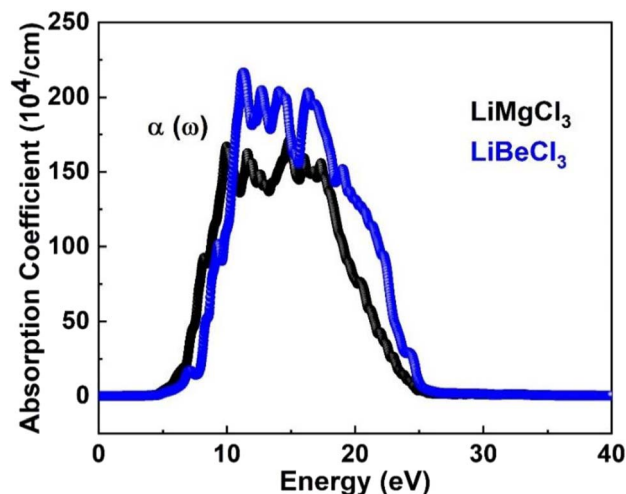


Fig. 7 The computed absorption coefficient for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.



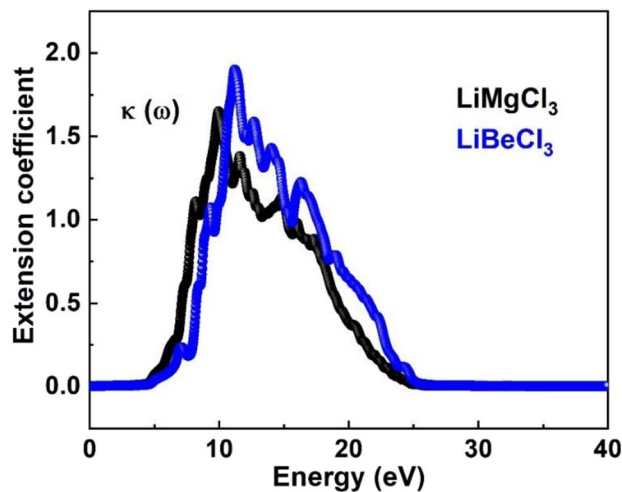


Fig. 8 The computed extinction co-efficient for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.

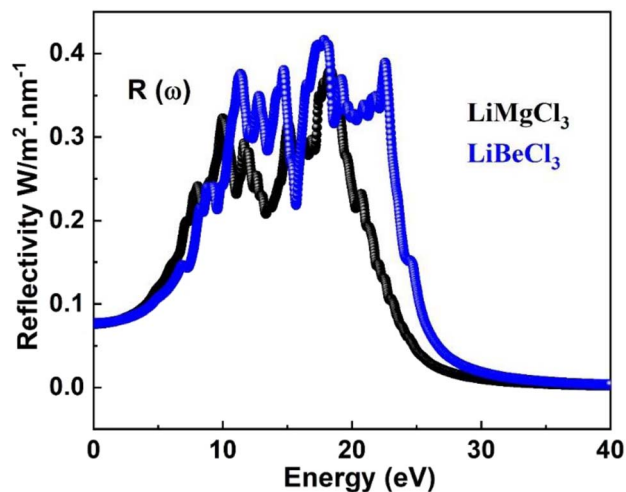


Fig. 10 The computed optical reflectivity for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.

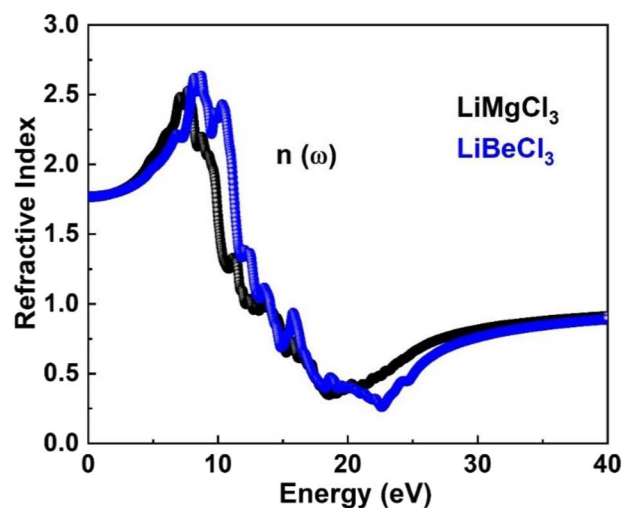


Fig. 9 The computed refractive index for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.

8 eV, while LiBeCl_3 reaches 2.8 at 9 eV. This indicates that when light passes through LiBeCl_3 , there is a greater delay due to electronic polarization compared to LiMgCl_3 , as evidenced by the higher refractive index of LiBeCl_3 . The size of the constituent atoms influences the degree of electronic polarization, with larger atoms leading to greater polarization, resulting in slower photon velocity within the material and a lower refractive index. For LiMgCl_3 and LiBeCl_3 , the refractive index decreases to 0.32 at 18 eV and 0.256 at 23.4 eV, respectively.

3.3.6. The optical reflectivity. Reflectivity refers to a material's ability to reflect energy from its surface. Fig. 10 shows the spectrum curves of $R(\omega)$ (reflectivity) for the LiRCl_3 halide perovskites ($R = \text{Be}$ and Mg), as determined by eqn (7). The static values of $R(0)$ for both LiRCl_3 compounds are 0.077. As the photon energy increases, the likelihood of photons being reflected by the material's surface also increases. The optical reflectivity of LiMgCl_3 and LiBeCl_3 both exhibit an upward trend

with increasing photon energy, as depicted in Fig. 9. In the photon energy range of 10 eV to 25 eV, LiBeCl_3 displays a higher maximum optical reflectivity of 0.417 compared to LiMgCl_3 , which reaches a maximum of 0.379. However, in the lower photon energy range of 0 eV to 5 eV, the optical reflectivity of both compounds remains relatively low.

3.3.7. The energy loss function. The energy loss function (ELF) quantifies the energy dissipated by an electron as it traverses a material, resulting from interactions with other electrons or the crystal lattice. It is derived from the imaginary part of the inverse dielectric function and provides valuable insights into intra-band, inter-band, and plasmonic phenomena within a material. Fig. 11 illustrates the ELF for both LiMgCl_3 and LiBeCl_3 , highlighting the distinct energy loss characteristics of each material. The plot reveals that the energy loss functions for both materials remain zero within the photon

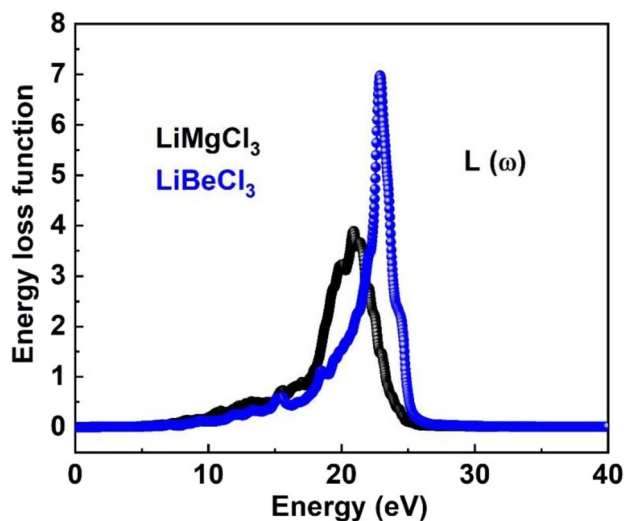


Fig. 11 The computed energy loss function for LiRCl_3 ($R = \text{Be}$ and Mg) halide perovskites.



energy range of 0 to 8.2 eV. However, as the photon energy increases, the ELF starts to rise, reaching a peak at 21.2 eV for LiMgCl₃ and 23.46 eV for LiBeCl₃. The energy loss function allows for an understanding of how a material responds to external electromagnetic disturbances and provides valuable insights into its properties. The observed optical properties in our findings align with the prior literature,^{2,5-7,57-66} which enhances their potential for optoelectronic applications.

3.4. The elastic properties

The measurement of elastic constants is crucial to investigate a material's elastic properties since they define how a material reacts to external forces and shed light on its mechanical properties. The mechanical stability and toughness of a material are thus revealed by such constants. The IRelast package is a computational tool used to calculate elastic constants for a cubic crystal with only three independent constants C_{11} , C_{12} , and C_{44} . When a material deforms in an elastic domain, these constants can be used to connect the mechanical response to the material's ductility or fragility. In this work, the elastic constants for the ternary LiRCl₃ halide perovskites compounds (R = Be and Mg) were examined using the GGA functional under typical cubic structure circumstances. Table 2 provides a summary of the researched elastic constants and other mechanical properties. The results demonstrate that both of the investigated chloroperovskite materials exhibit high compression resistance, as evidenced by their significant values of elastic constants obtained using the IRelast package for cubic crystals with only three independent constants C_{11} (65.35 GPa) in comparison to C_{12} (17.30 GPa) and C_{44} (14.47 GPa) for LiMgCl₃ and C_{11} (75.74 GPa) in comparison to C_{12} (29.73 GPa) and C_{44} (45.91 GPa) for LiBeCl₃, respectively. For both the compounds the examined elastic constants are positive which meets with the Born–Huang mechanical stability criteria⁶⁹ $C_{11} > 0$, $C_{12} > 0$, $C_{44} > 0$, $C_{11} + 2C_{12} > 0$, $C_{11} - C_{12} > 0$, and $B > 0$, a mechanically stable state for both of the chosen materials is predicted. To investigate the ductile or brittle behavior of a material, two essential parameters, namely the bulk modulus

“ B ” and the shear modulus “ G ”, were deduced. The estimations of Voigt and Reuss are commonly used to determine both “ B ” and “ G ” in general.⁷⁰ The compressibility and shear moduli, the Young's and Reuss approximations, the Poisson's ratio, and the anisotropy factor are given in the following formulas:

$$G = \frac{1}{2}(G_V + G_R) \quad (10)$$

$$G_V = \frac{1}{5}(C_{11} - C_{12} + 3C_{44}) \quad (11)$$

$$G_R = \frac{5C_{44}(C_{11} - C_{12})}{4C_{44} + 3(C_{11} - C_{12})} \quad (12)$$

$$E = \frac{9GB}{3B + G} \quad (13)$$

$$\nu = \frac{3B - 2G}{2(3B + G)} \quad (14)$$

$$A = 2C_{44}/(C_{11} - C_{12}) \quad (15)$$

In the equations above, the G , G_V , G_R , E , ν , and A represent the shear modulus, Voigt shear modulus, Reuss shear modulus, Young's modulus, Poisson's ratio, and anisotropy factor, in that order.

The bulk modulus (B), shear modulus (G), and Young's modulus (E) values obtained from the study can be utilized to assess the stiffness of the investigated material. According to the estimated values of B , E , and G , which are listed in Table 2, both of the ternary halide perovskites LiRCl₃ (R = Be and Mg) compounds are rigid and have a high scratch resistance. The Poisson's ratio “ ν ” is a crucial factor to consider when assessing a material's brittleness and malleability. LiMgCl₃ and LiBeCl₃ have investigated “ ν ” values of 0.26 and 0.18, respectively. This indicates that the chosen materials are brittle. The Pugh ratio, which defines whether a material is brittle or ductile, is another important factor. The “ B/G ” (Pugh ratio) cutoff value is 1.75.⁷¹ If the Pugh ratio is less than 1.75, a material will show brittle behavior, while a higher value indicates ductile behavior. The values of “ B/G ” for both materials, as in this paper, are 1.67 for LiMgCl₃ and 1.29 for LiBeCl₃ which is consistent with passion ratio confirming these materials to be brittle. Whether a material is isotropic or anisotropic is determined by the expected value of anisotropy factor “ A ,” whose critical value is “1”. Both of these halide perovskites are anisotropic based on the values calculated for them, which are 0.60 for LiMgCl₃ and 1.99 for LiBeCl₃. Another factor that determines how ductile a material is the Cauchy pressure. As shown in Table 2, the calculated Cauchy pressures for LiMgCl₃ and LiBeCl₃ are 2.83 GPa and –16.17 GPa, respectively, demonstrating brittle behavior for both of the materials (Table 3).

In summary, similar recent studies have been conducted to explore the structural, electronic, optical, and elastic properties of halide perovskites. It has been observed that our findings align with these studies, particularly in terms of structural and

Table 2 The investigated mechanical parameters which includes elastic constants C_{11} , C_{12} and C_{44} in GPa, bulk modulus B in GPa, anisotropy factor A , Poisson's ratio ν , Cauchy pressure $C_{12} - C_{44}$ in GPa, Young's modulus E in GPa, shear modulus G in GPa, and Pugh ratio B/G for LiMgCl₃ and LiBeCl₃ ternary chloroperovskites compounds

Mechanical parameters	LiMgCl ₃	LiBeCl ₃
C_{11} (GPa)	65.35	75.74
C_{12} (GPa)	17.30	29.73
C_{44} (GPa)	14.47	45.91
B (GPa)	33.36	45.15
A	0.603	1.996
ν	0.268	0.180
$C_{12} - C_{44}$ (GPa)	2.83	–16.17
E	46.40	86.719
G	17.75	34.79
B/G	1.67	1.29



Table 3 The effective masses of holes (m_h^*) and electrons (m_e^*) in LiBeCl₃ and LiMgCl₃

Compound	m_e^*/m_0	m_h^*/m_0	m_e^*/m_h^*
LiBeCl ₃	0.037	-0.114	-0.324
LiMgCl ₃	0.041	-0.125	-0.328

mechanical stability. Moreover, the significant electronic properties make these materials suitable for various applications. Additionally, their optical and electronic properties make them well-suited for utilization in a wide range of optoelectronic devices.

4. Conclusion

In conclusion, using the DFT embedded in WIEN2K, the focus of our investigation was on the structural and optoelectronic, mechanical and phonon properties of LiRCl₃ (R = Be and Mg), which are alkali metal-based halide perovskites materials. The stability of both LiRCl₃ (R = Be and Mg) halide perovskites at room temperature was demonstrated by fitting the energy–volume curve using the Birch–Murnaghan equation of state, as well as analyzing the tolerance factor and formation energy. The results showed that both materials are structurally and thermodynamically stable and maintain their geometries without undergoing any structural distortion. With indirect band gap energies of 4.02 and 4.07 eV for LiMgCl₃ and LiBeCl₃, respectively, the calculated electronic properties by exploiting the mBJ approach revealed the materials' insulating nature. The calculated densities of states (DOS) showed the same band gap energy values as that of band structure, confirming insulating nature for both the materials. These materials exhibit promising properties for use in UV devices, as demonstrated by their optical characteristics, including dielectric function, optical conductivity, reflectivity, refraction, and energy loss function. Additionally, the elastic properties of both materials indicate their mechanical stability. However, LiMgCl₃ displayed a weaker ability to withstand changes in volume compared to LiBeCl₃, which demonstrated a stronger ability to bear compressive and shear stresses. It was also found that none of the compounds are completely isotropic, and both LiMgCl₃ and LiBeCl₃ demonstrated ductile nature. Furthermore, the observed UV-Vis absorption and optical conductivity in both materials make them promising candidates for optoelectronic devices. Therefore, we hope that our findings will spark the interest of experimental studies for further investigations in this field.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors extend their appreciation to the Deanship of Scientific Research at King Khalid University Abha 61421, Asir,

Kingdom of Saudi Arabia for funding this work through the Large Groups Project under grant number RGP.2/142/44. The authors acknowledge the Deanship of Scientific Research, Vice Presidency for Graduate Studies and Scientific Research at King Faisal University, Saudi Arabia, for financial support under the annual funding track [GRANT3666].

References

- M. Husain, et al., Insight into the Structural, Mechanical and Optoelectronic Properties of Ternary Cubic Barium-Based BaMCl₃ (M= Ag, Cu) Chloroperovskites Compounds, *Crystals*, 2023, **13**(1), 140.
- M. Husain, et al., Examining computationally the structural, elastic, optical, and electronic properties of CaQCl₃ (Q= Li and K) chloroperovskites using DFT framework, *RSC Adv.*, 2022, **12**(50), 32338–32349.
- S. A. Khattak, et al., First-principles structural, elastic and optoelectronics study of sodium niobate and tantalate perovskites, *Sci. Rep.*, 2022, **12**(1), 21700.
- S. A. Khattak, et al., Investigation of Structural, Mechanical, Optoelectronic, and Thermoelectric Properties of BaXF₃ (X= Co, Ir) Fluoro-Perovskites: Promising Materials for Optoelectronic and Thermoelectric Applications, *ACS Omega*, 2023, **8**(6), 5274–5284.
- N. Rahman, et al., First-principles calculations to investigate structural, elastic, optical, and thermoelectric properties of narrow band gap semiconducting cubic ternary fluoroperovskites barium based BaMF₃ (M= Ag and Cu) compounds, *J. Mater. Res. Technol.*, 2022, **21**, 2168–2177.
- S. A. Shah, et al., Insight into the Structural, Electronic, Elastic, Optical, and Magnetic Properties of Cubic Fluoroperovskites ABF₃ (A= Tl, B= Nb, V) Compounds: Probed by DFT, *Materials*, 2022, **15**(16), 5684.
- S. A. Shah, et al., Insight into the exemplary structural, elastic, electronic and optical nature of GaBeCl₃ and InBeCl₃: a DFT study, *RSC Adv.*, 2022, **12**(13), 8172–8177.
- S. Körbel, M. A. L. Marques and S. Botti, Stability and electronic properties of new inorganic perovskites from high-throughput ab initio calculations, *J. Mater. Chem. C*, 2016, **4**(15), 3157–3167.
- N. Erum and M. A. Iqbal, Physical properties of fluorine based perovskites for vacuum-ultraviolet-transparent lens materials, *Chin. J. Phys.*, 2017, **55**(3), 893–903.
- H. Manaa, Y. Guyot and R. Moncorge, Spectroscopic and tunable laser properties of Co²⁺-doped single crystals, *Phys. Rev. B*, 1993, **48**(6), 3633.
- M. C. M. de Lucas, F. Rodriguez and M. Moreno, Excitation and emission thermal shifts in ABF₃: Mn²⁺ perovskites: coupling with impurity vibrational modes, *J. Phys.: Condens. Matter*, 1995, **7**(38), 7535.
- C. N. Avram, M. G. Brik, I. Tanaka and N. M. Avram, Electron–phonon interaction in the V²⁺: CsCaF₃ laser crystal: geometry of the [VF₆] 4– complex in the 4T_{2g} excited state, *J. Phys.: Condens. Matter*, 2005, **35**(1–4), 164–171.



- 13 J. M. Garcia-Lastra, J. Y. Buzare, M. T. Barriuso, J. A. Aramburu and M. Moreno, 3 d impurities in normal and inverted perovskites: Differences are not explained by ligand field theory, *Phys. Rev. B*, 2007, **75**(15), 155101.
- 14 S. Shahrokhi, et al., Emergence of ferroelectricity in halide perovskites, *Small Methods*, 2020, **4**(8), 2000149.
- 15 F. Lahoz, B. Villacampa and R. Alcalá, The tetragonal to orthorhombic structural phase transition in RbCaF₃ single crystals: Influence on the local environment of different nickel probes, *J. Phys. Chem. Solids*, 1997, **58**(6), 881–892.
- 16 B. Ghebouli, M. A. Ghebouli and M. Fatmi, Structural, elastic, electronic, optical and thermal properties of cubic perovskite CsCdF₃ under pressure effect, *Eur. Phys. J.: Appl. Phys.*, 2011, **53**(3), 30101.
- 17 R. R. Daniels, G. Margaritondo, R. A. Heaton and C. C. Lin, Experimental study of the electronic structure of KMg F₃, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1983, **27**(6), 3878.
- 18 O. D. I. Moseley, T. A. S. Doherty, R. Parmee, M. Anaya and S. D. Stranks, Halide perovskites scintillators: Unique promise and current limitations, *J. Mater. Chem. C*, 2021, **9**(35), 11588–11604.
- 19 P. Lecoq, Development of new scintillators for medical applications, *Nucl. Instrum. Methods Phys. Res., Sect. A*, 2016, **809**, 130–139.
- 20 J. Glodo, et al., New developments in scintillators for security applications, *Phys. Procedia*, 2017, **90**, 285–290.
- 21 S. K. Mathanker, P. R. Weckler and N. Wang, Terahertz (THz) applications in food and agriculture: A review, *Trans. ASABE*, 2013, **56**(3), 1213–1226.
- 22 M. Güdel and Y. Nazé, X-ray spectroscopy of stars, *Astron. Astrophys. Rev.*, 2009, **17**(3), 309–408.
- 23 N. Kawano, et al., Effect of organic moieties on the scintillation properties of organic–inorganic layered perovskite-type compounds, *Jpn. J. Appl. Phys.*, 2016, **55**(11), 110309.
- 24 M. N. Murshed, M. E. El Sayed, S. Naji and A. Samir, Electronic and optical properties and upper light yield estimation of new scintillating material TlMgCl₃: Ab initio study, *Results Phys.*, 2021, **29**, 104695.
- 25 S. Khan, S. U. Zaman, R. Ahmad, N. Mehmood, M. Arif and H. J. Kim, Ab initio investigations of structural, elastic, electronic and optical properties of the fluoroperovskite TlXF₃ (X= Ca, Cd, Hg, and Mg) compounds, *Mater. Res. Express*, 2020, **6**(12), 125923.
- 26 Q. V. Phan, H. J. Kim, G. Rooh and S. H. Kim, Tl₂ZrCl₆ crystal: Efficient scintillator for X- and γ -ray spectroscopies, *J. Alloys Compd.*, 2018, **766**, 326–330.
- 27 H. Lee, et al., Retarding crystallization during facile single coating of NaCl-incorporated precursor solution for efficient large-area uniform perovskite solar cells, *ACS Appl. Mater. Interfaces*, 2016, **8**(43), 29419–29426.
- 28 L. Dimesso, M. Wussler, T. Mayer, E. Mankel and W. Jaegermann, Inorganic alkali lead iodide semiconducting APbI₃ (A= Li, Na, K, Cs) and NH₄PbI₃ films prepared from solution: structure, morphology, and electronic structure, *AIMS Mater. Sci.*, 2016, **3**(3), 737–755.
- 29 Z. Tang, et al., Modulations of various alkali metal cations on organometal halide perovskites and their influence on photovoltaic performance, *Nano Energy*, 2018, **45**, 184–192.
- 30 A. Suzuki and T. Oku, First-principles calculation study of electronic structures of alkali metals (Li, K, Na and Rb)-incorporated formamidinium lead halide perovskite compounds, *Appl. Surf. Sci.*, 2019, **483**, 912–921.
- 31 Y. H. Park, et al., Inorganic rubidium cation as an enhancer for photovoltaic performance and moisture stability of HC (NH₂)₂PbI₃ perovskite solar cells, *Adv. Funct. Mater.*, 2017, **27**(16), 1605988.
- 32 A. Mishra, et al., Enhanced Operational Stability of Perovskite Light-Emitting Electrochemical Cells Leveraging Ionic Additives, *Adv. Opt. Mater.*, 2020, **8**(13), 2000226.
- 33 T. Liu, et al., High-performance formamidinium-based perovskite solar cells via microstructure-mediated δ -to- α phase transformation, *Chem. Mater.*, 2017, **29**(7), 3246–3250.
- 34 O. A. Syzgantseva, M. Saliba, M. Grätzel and U. Rothlisberger, Stabilization of the perovskite phase of formamidinium lead triiodide by methylammonium, Cs, and/or Rb doping, *J. Phys. Chem. Lett.*, 2017, **8**(6), 1191–1196.
- 35 R. B. dos Santos, R. Rivelino, G. K. Gueorguiev and A. Kakanakova-Georgieva, Exploring 2D structures of indium oxide of different stoichiometry, *CrystEngComm*, 2021, **23**(38), 6661–6667.
- 36 C. Lundgren, A. Kakanakova-Georgieva and G. K. Gueorguiev, A perspective on thermal stability and mechanical properties of 2D Indium Bismide from ab initio molecular dynamics, *Nanotechnology*, 2022, **33**(33), 335706.
- 37 G. K. Gueorguiev, S. Stafström and L. Hultman, Nano-wire formation by self-assembly of silicon–metal cage-like molecules, *Chem. Phys. Lett.*, 2008, **458**(1–3), 170–174.
- 38 C. D. Wick and L. X. Dang, Recent advances in understanding transfer ions across aqueous interfaces, *Chem. Phys. Lett.*, 2008, **458**(1–3), 1–5.
- 39 Z. Xiao, Z. Song and Y. Yan, From lead halide perovskites to lead-free metal halide perovskites and perovskite derivatives, *Adv. Mater.*, 2019, **31**(47), 1803792.
- 40 M. Ameri, F. Bennar, S. Amel, I. Ameri, Y. Al-Douri and D. Varshney, Structural, elastic, thermodynamic and electronic properties of LuX (X= N, Bi and Sb) compounds: first principles calculations, *Phase Transitions*, 2016, **89**(12), 1236–1252.
- 41 L. J. Sham and W. Kohn, One-particle properties of an inhomogeneous interacting electron gas, *Phys. Rev.*, 1966, **145**(2), 561.
- 42 P. Blaha, K. Schwarz, G. K. H. Madsen, D. Kvasnicka and J. Luitz, *WIEN2k: An Augmented Plane Wave plus Local Orbitals Program for Calculating Crystal Properties*, Technische Universität Wien, Wien, 2001.
- 43 J. Tao, J. P. Perdew, H. Tang and C. Shahi, Origin of the size-dependence of the equilibrium van der Waals binding between nanostructures, *J. Chem. Phys.*, 2018, **148**(7), 74110.
- 44 M. Jamal, M. Bilal, I. Ahmad and S. Jalali-Asadabadi, IRelast package, *J. Alloys Compd.*, 2018, **735**, 569–579.



- 45 D. Koller, F. Tran and P. Blaha, Merits and limits of the modified Becke-Johnson exchange potential, *Phys. Rev. B*, 2011, **83**(19), 195134.
- 46 E. Sakalauskas, et al., Dielectric function and optical properties of Al-rich AlInN alloys pseudomorphically grown on GaN, *J. Phys. D: Appl. Phys.*, 2010, **43**(36), 365102.
- 47 A. Kokalj, XCrySDen—a new program for displaying crystalline structures and electron densities, *J. Mol. Graphics Modell.*, 1999, **17**(3–4), 176–179.
- 48 A. Vaught, Graphing with Gnuplot and Xmgr: two graphing packages available under linux, *Linux J.*, 1996, **1996**(28es), 7.
- 49 P. M. Edwards, Origin 7.0: scientific graphing and data analysis software, *J. Chem. Inf. Comput. Sci.*, 2002, **42**(5), 1270–1271.
- 50 H. Wainer, Visual Revelations: Elegance, Grace, Impact, and Graphical Displays, *Chance*, 1991, **4**(4), 45–47.
- 51 H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Phys. Rev. B*, 1976, **13**(12), 5188.
- 52 F. D. Murnaghan, The compressibility of media under extreme pressures, *Proc. Natl. Acad. Sci. U. S. A.*, 1944, **30**(9), 244.
- 53 A. A. Emery and C. Wolverton, High-throughput DFT calculations of formation energy, stability and oxygen vacancy formation energy of ABO₃ perovskites, *Sci. Data*, 2017, **4**(1), 1–10.
- 54 G. Nazir, et al., Tuning of band gap by anions (Cl, Br, I) of double perovskites Rb₂AgAsX₆ (Cl, Br, I) for solar cells and thermoelectric applications, *Phys. Scr.*, 2023, 025811.
- 55 W.-H. Liu, et al., Predicting impact sensitivity of energetic materials: insights from energy transfer of carriers, *Acta Mater.*, 2022, **236**, 118137.
- 56 S. Bouhmaidi, A. Marjaoui, A. Talbi, M. Zanouni, K. Nouneh and L. Setti, A DFT study of electronic, optical and thermoelectric properties of Ge-halide perovskites CsGeX₃ (X= F, Cl and Br), *Comput. Condens. Matter*, 2022, e00663.
- 57 N. Rahman, et al., First principle study of structural, electronic, optical and mechanical properties of cubic fluoro-perovskites:(CdXF₃, X= Y, Bi), *Eur. Phys. J. Plus*, 2021, **136**(3), 1–11.
- 58 A. Habib, et al., Insight into the Exemplary Physical Properties of Zn-Based Fluoroperovskite Compounds XZnF₃ (X= Al, Cs, Ga, In) Employing Accurate GGA Approach: A First-Principles Study, *Materials*, 2022, **15**(7), 2669.
- 59 M. Husain, et al., First Principle Study Of The Structural, Electronic, And Mechanical Properties Of Cubic Fluoroperovskites:(ZnXF₃, X= Y, Bi), *Fluoride*, 2020, **53**(4), 657–667.
- 60 M. Sohail, et al., First-principal investigations of electronic, structural, elastic and optical properties of the fluoroperovskite TILF₃ (L= Ca, Cd) compounds for optoelectronic applications, *RSC Adv.*, 2022, **12**(12), 7002–7008.
- 61 H. Khan, M. Sohail, N. Rahman, M. Hussain, A. Khan and H. H. Hegazy, Theoretical study of different aspects of Al-based Fluoroperovskite AlMF₃ (M= Cu, Mn) compounds using TB-MBJ potential approximation method for generation of energy, *Results Phys.*, 2022, 105982.
- 62 M. Husain, et al., Structural, electronic, elastic, and magnetic properties of NaQF₃ (Q= ag, Pb, Rh, and Ru) fluoroperovskites: A first-principle outcomes, *Int. J. Energy Res.*, 2022, **46**(3), 2446–2453.
- 63 J. Saddique, et al., Modeling structural, elastic, electronic and optical properties of ternary cubic barium based fluoroperovskites MBaF₃ (M= Ga and In) compounds based on DFT, *Mater. Sci. Semicond. Process.*, 2022, **139**, 106345.
- 64 N. Rahman, et al., First principle study of structural, electronic, optical and mechanical properties of cubic fluoro-perovskites:(CdXF₃, X= Y, Bi), *Eur. Phys. J. Plus*, 2021, **136**(3), 1–11.
- 65 M. Husain, et al., Exploring the exemplary structural, electronic, optical, and elastic nature of inorganic ternary cubic XBaF₃ (X= Al and Tl) employing the accurate TB-mBJ approach, *Semicond. Sci. Technol.*, 2022, **37**(7), 75004.
- 66 S. Khawar, et al., First-principles calculations to investigate structural, electronic, optical, and magnetic properties of a scintillating double perovskite halide (Cs₂LiCeCl₆), *J. Mater. Res. Technol.*, 2022, **21**, 4790–4798.
- 67 Q. Mahmood, et al., Study of new lead-free double perovskites halides Tl₂TiX₆ (X= Cl, Br, I) for solar cells and renewable energy devices, *J. Solid State Chem.*, 2022, **308**, 122887.
- 68 M. A. Khan, H. A. Alburaih, N. A. Noor and A. Dahshan, Comprehensive investigation of Opto-electronic and transport properties of Cs₂ScAgX₆ (X= Cl, Br, I) for solar cells and thermoelectric applications, *Sol. Energy*, 2021, **225**, 122–128.
- 69 M. Born, On the stability of crystal lattices. I, *Math. Proc. Cambridge Philos. Soc.*, 1940, **36**(2), 160–172.
- 70 R. Hill, The elastic behaviour of a crystalline aggregate, *Proc. Phys. Soc. Sect. A*, 1952, **65**(5), 349.
- 71 S. F. Pugh, XCII. Relations between the elastic moduli and the plastic properties of polycrystalline pure metals, *London, Edinburgh Dublin Philos. Mag. J. Sci.*, 1954, **45**(367), 823–843.

