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Targeted recovery of metals from thermoelectric generators (TEGs) using chloride brines and ultrasound†

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Recovery of elemental copper, bismuth, tellurium, antimony and tin from thermoelectric generators (TEGs) is vital to recover the high content of critical metals and potential risk of environmental pollution as a result of incorrect disposal of TEGs and to enable the circular economy. In this work, aqueous choline chloride and calcium chloride hexahydrate brines were characterised and used in combination with copper(II) as an oxidising agent to leach copper and tin from TEGs. This permitted the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs to be readily separated from the ceramic substrates by filtration. It was shown that at low chloride content, surface passivation and solubility of the oxidised species were the limiting factors towards oxidation, whereas solvent viscosity (mass transport) was the limiting factor at high chloride content. The copper(II) species formed in the different brines were determined *via* UV-vis spectroscopy. The redox potentials of the oxidising species were found to be significantly altered by choline chloride content, but not so much by calcium chloride hexahydrate content, suggesting variation in chloride activity within the different brines. The developed approach has been shown to be a viable and scalable method to recover high value critical metals from e-waste containing TEGs.

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Sustainability spotlight

Thermoelectric generators are sustainable and cost-effective devices that can convert waste heat into additional electrical power. The global demand for bismuth and tellurium is forecast to increase, hence the recycling of old devices is becoming more important. The sustainable advancement of the present work is the selective recovery of bismuth tellurides from scrap thermoelectric devices *via* the use of copper(II) as a mild oxidising agent in aqueous chloride brines. This work falls under Goal 12 “Responsible consumption and production”, as it describes a method for enabling quick and clean recovery of the critical semiconductor elements, which in turn could reduce the current reliance on natural resources to supply demand for these devices.

1 Introduction

Energy efficiency is a vital topic and it has been estimated that about 75% of the energy produced worldwide dissipates as heat into the environment.¹ Thermoelectric generators (TEGs, also called Seebeck generators) are sustainable and cost-effective, although inefficient, devices that have demonstrated promising results to convert waste heat into additional electrical power.² TEGs are widely used in many applications, such as power generators – especially in automobiles, thermal energy sensors, for cooling in electronics and photonics, and for

medical treatment, among others,³ resulting in increased global demand for the key elements.^{4–7} Interest in TEGs has grown rapidly in recent years in order to harvest energy for IoT devices such as sensors and actuators.⁸ TEGs have a simple design, no moving parts, a long lifespan and require low maintenance enabling their application in numerous situations. The most common modules are based of bismuth telluride (Bi_2Te_3), doped with antimony to produce either a p-type ($(\text{Sb}_{0.8}\text{Bi}_{0.2})_2\text{Te}_3$) or n-type ($\text{Bi}_2(\text{Te}_{0.8}\text{Se}_{0.2})_3$) materials.⁹ The scarcity of these elements has limited the development of Bi_2Te_3 based TEGs.^{10,11} Alternative materials, such as skutterudites, Ca/Mg oxides, and Mg_2Si , along with the discovery of new materials such as Heusler compounds (magnetic intermetallic compounds), has allowed the development of potentially more cost-effective TEGs,¹² although such devices have not yet proven their commercial success.

Bi is listed as Critical Raw Material by the EU in 2020 owing to its scarcity and supply chain risks, whilst Te and Se are

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amongst the least abundant elements on the Earth's crust.¹³ All three tend to be sourced as by-products of other metal processing,^{5,14,15} and are therefore vulnerable to supply chain risks. Hence, improved recovery of both bismuth, tellurium and antimony from end-of-life TEG devices has an increasing importance.

Current recycling processes for TEGs are relatively under studied. Mechanical cutting and milling of the thermoelectric materials has been used to liberate different fractions, which can then be size separated.¹⁶ However, processes reported so far result in dissolution of the TEGs or reduction of their size, therefore re-processing is required to remanufacture the TEG from the obtained recycled materials. Some hydrometallurgical methods for recycling TEGs have been reported in the literature, such as non-selective leaching with hydrochloric, nitric, or sulphuric acids, followed by the use of hydroxide and hydrazine to precipitate the bismuth and tellurium powders, or directly regenerate new forms of bismuth telluride.^{17–19} The bacterial recovery of tellurium in both solid and gaseous forms has also been attempted.²⁰ These approaches require further processing of the leaching solution to recover the metals.

An alternative method for recovery of the thermoelectric legs from waste TEGs would be to selectively leach out the base metal layers that adhere the thermoelectric materials to the ceramic substrates. Solvents such as deep eutectic solvents (DESS) and concentrated chloride brines have the advantage of high chloride contents, which leads to different speciation and redox behaviour, hence allowing to selectively dissolve target metals. This approach has been applied successfully to the selective leaching of metals from minerals, dissolution of metal oxides from end-of-life batteries, leaching of silver and aluminium from solar cells and copper from circuit boards, among others.^{21–24} In addition to DESS, chloride brines have previously been used as solvents for the leaching of metals from minerals,^{25–28} and the leaching of silver and aluminium from solar panels,²⁴ using a number of different oxidising agents, including Cu^{II} and Fe^{III}.

In this work, choline chloride (ChCl) and calcium chloride hexahydrate (CaCl₂·6H₂O) brines have been applied for selective dissolution of copper from thermoelectric devices using copper(II) chloride as the oxidising agent to liberate the Bi_{2–x}Sb_xTe₃ legs from the ceramic substrate. Copper(II) was selected as the oxidising agent to minimise impurity contamination because the Cu^{II/I} redox potential is less anodic relative to the target metals, resulting in more noble metals remaining unaffected. This allows for easy recovery of the individual thermoelectric legs *via* simple filtration. Both materials used in this process are abundantly available; CaCl₂·6H₂O is a waste by-product from the Solvay process,²⁹ and ChCl is a bulk commodity chemical, which has no additional REACH requirements. The use of these materials enables a rapidly scalable recycling process. The copper(II) species present in solution are characterised along with their redox properties as a way to understand the relative oxidising ability towards the target elements. Chloride concentration (and activity) will likely be different between the organic and inorganic chloride sources, affecting the soluble metal species formed. This method

was compared with high-intensity ultrasonication, which allows efficient removal of the brittle Bi_{2–x}Sb_xTe₃ legs from the ceramic substrate.

2 Materials and methods

2.1 Reagents and solvent preparation

The solvents used in this study are prepared by dissolving choline chloride (ChCl, Sigma Aldrich, >98%) or calcium chloride hexahydrate (CaCl₂·6H₂O, Honeywell, >97%) in deionised water (Elga Purelab Option apparatus) at varying molar ratios of salt-to-water (1:3, 1:4, 1:10, and 1:50). The mixtures were stirred at 50 °C until a clear and homogeneous liquid was obtained.

The leaching solutions were prepared by dissolving 0.1 mol dm^{–3} anhydrous copper(II) chloride (Acros Organics, 99%) in the appropriate solvent. The solutions for voltammetric analysis were prepared by dissolving 0.02 mol dm^{–3} of anhydrous copper(II) chloride (Alfa Aesar, 99.9%), tin(II) chloride (Alfa Aesar, 98%), tellurium(IV) chloride (Sigma Aldrich, 99%), antimony(III) chloride (Sigma Aldrich, 99%) and bismuth(III) chloride (Alfa Aesar, 98%) in the different brines. The UV-vis spectra were recorded using the same solutions. The elements investigated have a concentration higher than 1 wt% in the thermoelectric materials.

2.2 Leaching studies

Metal leaching was carried out by placing *ca.* 30 mg of a Cu foil, or Sn wire, Sb lump, Te or Bi chunk in 5 mL of the appropriate solvent containing the oxidising agent (0.1 mol dm^{–3}). Without an oxidising agent, the solvent alone is unable to dissolve the metals (no leaching was observed for any of the metals). The system was stirred (200 rpm) at 50 °C for 1 hour, before the remaining solid metal was removed, rinsed with deionised water, air-dried, and weighed to determine the mass of metal dissolved. Table 1 shows the salt-to-water ratio of the brines selected together with their theoretical water and chloride concentration. Variation in the water content of brines (due to purity of the chemical sources, losses during heating...) is unlikely to affect the chemistry of the solutes since brines already contain a large amount of water. The values are compared with DESS using ethylene glycol as an HBD, which provide a similar chloride content.

Thermoelectric materials were provided by European Thermodynamics Ltd, with a schematic of their structure shown in Fig. S1.† They are composed of two layers of Al₂O₃ ceramic wafer coated with a Cu conductor strip of 8 mm² size with 380 μm thickness. Between the Cu strip and the ceramic, an 8 mm² sized and 3 μm thick W interface layer is used to complete the thermal junction and attach to the underlying solder. A 300 μm thick layer of solder containing mostly Sn, but also Zn coats the Cu strips. Finally, two 1 mm³ thermoelectric legs of bismuth–antimony telluride or bismuth telluride doped with selenium (hereafter referred to as Bi_{2–x}Sb_xTe₃) are sandwiched between these solder layers to complete the device. Overall, the ceramic plate represents around 40 wt% of the thermoelectric materials,



Table 1 Calculated theoretical chloride and water concentrations in mol kg⁻¹ of brine. Viscosity measured at 25 °C

Brine/molar ratio	Chloride conc./mol kg ⁻¹	Water conc./mol kg ⁻¹	Viscosity/mPa s
CaCl₂·6H₂O brines			
CaCl ₂ ·6H ₂ O : 3H ₂ O	7.3	33.0	7.72(8)
CaCl ₂ ·6H ₂ O : 4H ₂ O	6.9	34.4	6.11(5)
CaCl ₂ ·6H ₂ O : 10H ₂ O	5.0	40.1	2.58(3)
CaCl ₂ ·6H ₂ O : 50H ₂ O	1.8	50.0	1.26(7)
ChCl brines			
ChCl : 3H ₂ O	5.2	15.5	14.08(6)
ChCl : 4H ₂ O	4.7	18.9	8.0(1)
ChCl : 10H ₂ O	3.1	31.3	4.56(2)
ChCl : 50H ₂ O	1.0	48.1	1.327(5)
Deep eutectic solvents			
ChCl : 2EG ^a	3.8	7.6 (EG)	37
CaCl ₂ ·6H ₂ O : 2EG ^b	5.8	23.3 (EG + H ₂ O)	59.48(8)

^a Data taken from D'Agostino *et al.*³⁰ ^b Data taken from Hartley *et al.*³¹

while soldering and junction elements other than Sn represent less than 1 wt% of the materials. The composition of the thermoelectric materials, obtained after aqua regia (a 3 : 1 volume ratio of hydrochloric and nitric acid) digestion of 100 mg of materials in 10 mL and analysis of the solution with ICP-MS is displayed in Table 2. The composition reported is in accordance with the composition of European thermodynamics' products.

During leaching experiments, a sample of thermoelectric material (around 1 × 0.5 cm) was immersed in different leaching solutions containing 0.1 mol dm⁻³ of Cu(II) chloride as an oxidising agent (the liquid to solid ratio is maintained at 20 mL g⁻¹). The solution was then ultrasonicated at room temperature for 30 minutes with a Fisherbrand FB15055 ultrasonic bath delivering an ultrasonic frequency of 37 kHz, a power supply of 150 W, and a power density of around 0.05 W cm⁻³. The sample was then removed from the solution, rinsed with deionised water, air-dried, and weighed. The metal content in the solution is estimated by ICP-MS after filtration with 0.2 µm PET syringe filters. The Bi_{2-x}Sb_xTe₃ legs are filtered from the solution with filter paper, rinsed with deionised water, air-dried, and weighed.

Table 2 Composition of the thermoelectric materials

Metal	Composition, wt%
Cu	21.3
Te	18.6
Bi	13.9
Sb	3.5
Sn	3.5
Ni	0.9
W	0.8
Se	0.5
Others (ceramic plate, Zn)	37.0

High intensity ultrasound experiments were carried out using a commercial ultrasonic system (Branson Sonics, 1.25DCXa20-V), using a 20 mm diameter cylinder sonotrode. The system operates at 20 kHz, with power variable up to 1250 W, and a maximum power intensity of 398 W cm². The two ceramic plates were manually separated first, allowing for a closer distance between the sonotrode and the thermoelectric material. During high intensity ultrasound experiments, a sample of thermoelectric material (around 1 × 1.5 cm) was submerged under deionised water (20 mL) and placed under the sonotrode at a distance of 1 mm. Ultrasound was applied to the surface at 50% power intensity (198 W cm⁻²) for 15 seconds. The resultant solids were collected *via* filtration, and rinsed with deionised water.

2.3 Instrumentation

The viscosity values of the different brines were measured at 25 °C using a Seiko EG&G QCM922A Quartz Crystal Microbalance (QCM), according to previously described procedures (Hartley *et al.*, 2022). Solvent densities were estimated by weighing 100 mL of the solvent heated at 25 °C in a 100 mL volumetric flask, using a 4 d.p. balance (Mettler Toledo). Conductivity values were recorded using a commercial chemical-resistant type conductivity probe (SC72SN-31-AA; Yokogawa) and metre (SC72 Personal Conductivity Meter; Yokogawa), with an integrated temperature sensor (accuracy within ± 0.7 °C). The cell constant for the commercial conductivity probe was 5.2 ± 0.5 cm⁻¹, and conductivity standards (certified traceable to NIST; VWR) of 44 479, 11 419, 8863, 1249, 885, 442, and 74 µS cm⁻¹ (at 19 °C) were used as received to perform calibrations. The reported conductivity values are an average of at least three repeat measurements.

The metal content and Bi : Sb ratio of the thermoelectric devices was obtained *via* ICP-MS with a Thermo Scientific iCAP-Qc apparatus. Digestion of a sample containing two of the Cu strip-BiTe legs was done in aqua regia. These solutions were filtered with 0.2 µm PET syringe filters before being diluted (1000 fold) with a 2% trace metal grade nitric acid (Fisher Scientific, 70%) solution. Lanthanum and rhenium are used as internal standard (10 µg dm⁻³ of each). Concentrations were obtained from calibration curves between 0.1 and 1 µg dm⁻³ for Te, 10 and 500 µg dm⁻³ for Sb and Bi, and 25 and 5000 µg dm⁻³ for Cu.

Absorption spectra were recorded with a Mettler Toledo UV5 Bio spectrometer, using quartz cuvettes with a path length of 10 mm or 0.1 mm, depending on the strength of the absorptions. To determine the composition of the materials, Scanning Electron Microscopy (SEM) and Energy Dispersive X-ray (EDX) images were obtained on a Field Emission Gun Scanning Electron Microscopy (Quanta 650 FEG-SEM apparatus) with an accelerating voltage of 20 kV.

The electrochemical behaviour of the target metals was studied by cyclic voltammetry (CV), using an IVIUMnSTAT potentiostat. A 0.5 mm diameter platinum disk was used as the working electrode, with a platinum flag counter electrode, and a 3 mol dm⁻³ KCl Ag/AgCl reference electrode. Prior to each



experiment, the working electrode was polished with a 0.3 μm and 0.05 μm alumina slurry, rinsed with deionised water, and air dried. The reference electrode potential was calibrated to the $[\text{Fe}(\text{CN})_6]^{3-/4-}$ redox couple, which is used as an internal standard to ensure comparability between the different solvents. The scan rate was 20 mV s^{-1} .

3 Results and discussion

The first section of this study investigates the electrochemical properties of the constituent elements in a TEG and compared these with the leaching solutions used in two brines, one with a high charge density (Ca^{2+}) and one with a lower charge density (choline). The second half of the study investigated the application of ultrasound to recycling of TEG devices.

3.1 Speciation of Cu(II) and the thermoelectric leg elements in the brines

In order to understand the ability of the selected oxidising agent to oxidise other metallic species, identification of metal speciation in the brines is essential. The speciation of Cu^{II} in the different brines is discussed in the ESI.† The absorption spectra of $\text{Cu}(\text{II})$ chloride in ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines with salt-to-water ratios of 1:50; 1:10; 1:4 and 1:3 are shown in Fig. S2(a) and (b),† and photographs of these systems are shown in Fig. S2(c).† The absorbance maxima for these spectra and predicted species are presented in Table S1,† while the plot of the molar extinction coefficient at 402 nm as a function of water content in the brines is displayed in Fig. S3.† These spectra indicate the presence of $[\text{CuCl}_4]^{2-}$ in the most concentrated choline chloride brines ($[\text{Cl}^-] \geq 4.7 \text{ mol kg}^{-1}$), mixed chloride-water species at lower chloride concentrations ($[\text{CuCl}(\text{H}_2\text{O})_n]^+$ and $[\text{CuCl}_3(\text{H}_2\text{O})_n]^-$) while no chloride species are present at the lowest chloride concentration ($[\text{CuCl}(\text{H}_2\text{O})_n]^+$ and $[\text{Cu}(\text{H}_2\text{O})_n]^{2+}$). In the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines, mixed chloride-aqua complexes are found at different chloride concentrations.

Absorption spectroscopy of the $\text{Bi}(\text{III})$, $\text{Te}(\text{IV})$, $\text{Sb}(\text{III})$ and $\text{Sn}(\text{II})$ chloride salts dissolved in different brines was used to determine the species present (Fig. S4,† absorbance maxima and proposed species are displayed in Table 3). In all eight of the different brines, Bi^{III} ions were present as chloride species, as shown by absorbance maxima present at *ca.* 210, 225, and 330 nm.^{32–34} There is a minor solvatochromic shift between the ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines (*ca.* 3 nm), which is much smaller than was seen for $\text{Cu}(\text{II})$ ions. Literature suggests the presence of $[\text{BiCl}_6]^{3-}$ in $\text{ChCl} : 2\text{EG}$, although multiple species were not ruled out.³⁵ In concentrated acidic brines (11 mol per kg chloride), EXAFS spectroscopy indicated an average coordination of 5–6 chloride ligands, whereas at low chloride content (1 mol kg^{-1}) an average of *ca.* 3–4 chloride ligands were coordinated, probably in the form of a mixture³⁶ of $[\text{BiCl}_2]^+$ and $[\text{BiCl}_5]^{2-}$. Therefore, the most likely species present in these ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines will be a mixture of 5–6 chloride coordination, except for the ChCl brine with a chloride content of 1 mol kg^{-1} , where a lower coordination number is expected.

Table 3 UV-vis absorbance maxima of the Bi^{III} , Sb^{III} , and Sn^{II} species in the different chloride brines, along with the predicted species

Solvent/chloride content	Absorbances/nm	Predicted species
Bi		
$\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines		
$[\text{Cl}^-] \geq 1.8 \text{ mol kg}^{-1}$	210, 222, 328	$[\text{BiCl}_5]^{2-}/[\text{BiCl}_6]^{3-}$
ChCl brines		
$[\text{Cl}^-] \geq 3.1 \text{ mol kg}^{-1}$	211, 224, 329	$[\text{BiCl}_5]^{2-}/[\text{BiCl}_6]^{3-}$
$[\text{Cl}^-] = 1.0 \text{ mol kg}^{-1}$	209, 222, 324	$[\text{BiCl}_2]^+$ and $[\text{BiCl}_3]^0$
Sb		
$\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines		
$[\text{Cl}^-] \geq 6.9 \text{ mol kg}^{-1}$	229, 264, 289	$[\text{SbCl}_4]^-$
$[\text{Cl}^-] = 5.0 \text{ mol kg}^{-1}$	219, 263, 280	$[\text{SbCl}_2]^+$ and $[\text{SbCl}_3]^0$
ChCl brines		
$[\text{Cl}^-] \geq 4.7 \text{ mol kg}^{-1}$	226, 241, 266, 285	$[\text{SbCl}_4]^-$
Sn		
$\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines		
$[\text{Cl}^-] \geq 1.8 \text{ mol kg}^{-1}$	220	$[\text{SnCl}_4]^{2-}$
ChCl brines		
$[\text{Cl}^-] \geq 1.0 \text{ mol kg}^{-1}$	222	$[\text{SnCl}_4]^{2-}$

The spectra for TeCl_4 dissolved in the different brines show a broad absorbance maximum below 200 nm, hence it is not possible to determine the exact species present *via* this technique. However, it is known that $\text{Te}(\text{IV})$ ions can form mixed ligand complexes, depending on chloride content. For example, in highly concentrated aqueous brines (13–14 mol per kg chloride) there is the possibility for $[\text{TeCl}_4]^-$ or $[\text{TeO}_{0.3}\text{Cl}_{3.6}]^0$ to form, where the oxygen may be from oxide, hydroxide, or water. However, in low chloride systems ($<5 \text{ mol kg}^{-1}$) the species are fully oxygen-coordinated. In $\text{ChCl} : 2\text{EG}$ a solution of TeCl_4 displays absorbance maxima at 248 and 289 nm,³² which is potentially related to $[\text{TeCl}_6]^{2-}$. Hence, the most likely species present in the ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines will be mixed ligand complexes.

The spectra for $\text{Sb}(\text{III})$ are more complex, as multiple absorbance maxima are present. In the ChCl brines there are four maxima present in the region from 200 to 350 nm. These are less distinct in the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines. UV-vis spectra of $[\text{SbCl}_4]^-$ in acetonitrile show two main absorbance regions at *ca.* 230–260 nm and *ca.* 280 nm, whereas the spectrum of $[\text{SbCl}_6]^{3-}$ displays absorbance maxima at *ca.* 240–270 nm and *ca.* 310 nm.³⁴ It is therefore most likely that in the brines with a ChCl concentration of 4.7 mol kg^{-1} and higher presented here, that the most likely species is $[\text{SbCl}_4]^-$. In the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brine with a chloride content of 5.0 mol kg^{-1} , another species is present, perhaps relating to a mixture of $[\text{SbCl}_2]^+$ and $[\text{SbCl}_3]^0$, as observed in acidic chloride brines containing *ca.* 3.5 mol kg per chloride.³⁷ For the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ and ChCl brines with the lowest chloride content, Sb was found to precipitate instantaneously, possibly as insoluble antimony oxychloride species (SbOCl), since antimony is known to hydrolyse quickly when diluted in water.³⁸ Therefore, Sb behaviour in ChCl brine having a chloride content lower than 4.7 mol kg^{-1} as well as $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brine having a chloride content of 1.8 mol kg^{-1} was not investigated further.



The absorbance maxima for the Sn(II) complexes vary from 222 nm in the highest chloride concentration systems, down to 209 nm in the least concentrated systems. Sn(II) is known to form $[\text{SnCl}_3]^-$ and $[\text{SnCl}_4]^{2-}$ complexes in aqueous media, even at relatively low chloride contents ($0.35 \text{ mol dm}^{-3} \text{ HCl}^{39}$). In DES media, $[\text{SnCl}_4]^{2-}$ complexes are formed.⁴⁰ Therefore, it is likely that the majority species in these chloride brines will be $[\text{SnCl}_4]^{2-}$, with the shift in absorbance maxima values being related to the change in ionic strength and solvation environment of the different solvents. There is minimal solvatochromic shift between the ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines.

3.2 Electrochemistry and leaching

3.2.1 Electrochemistry of $\text{Cu}^{\text{II/I}}$ and the thermoelectrics in the brines. Voltammetry of the target elements in the different brines were recorded to identify whether oxidation with the different Cu^{II} species present will be thermodynamically viable. Redox potentials as a function of chloride content are presented in Fig. 1, with the corresponding CVs shown in Fig. S5† (Cu) and Fig. S6† (Sb, Bi, Te, Sn).

The formal electrode potential of the $\text{Cu}^{\text{II/I}}$ couple was measured in the different brines and it was found that the presence of different Cu species resulted in a cathodic shift between most and least concentrated chloride systems of up to 0.3 V in the ChCl brines, and of up to 0.05 V in the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines. Increasing the chloride content causes the formal electrode potential to increase. This means that the Cu^{II} species present in the systems with the highest water contents are less oxidising relative to the $[\text{Fe}(\text{CN})_6]^{3-/4-}$ couple compared to those in the highest chloride content systems. This effect can be used to tune the oxidising strength of Cu^{II} towards other metals, assuming that the formal electrode potentials of the target metals are not altered adversely by the changing solvent composition. Interestingly, the redox potential of the $\text{Cu}^{\text{II/I}}$ couple is consistently more than 250 mV more cathodic in $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines compared to ChCl brines having the same

salt-to-water ratios, regardless of the actual chloride content. It should be noted that the redox potential of the $\text{Cu}^{\text{II/I}}$ couple in $\text{ChCl} : 2\text{EG}$ is of a similar value to that of the ChCl brines containing $>3 \text{ mol per kg}$ chloride.

The CVs of Bi chloride in the various brines show two redox events, the first being related to the $\text{Bi}^{3+/0}$ couple (at around $-0.6 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$ in $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines and around $-0.4 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$ in ChCl brines) while the other is related to the formation of platinum–bismuth alloys,⁴¹ and the subsequent oxidation of Bi from the alloy at more anodic potentials.⁴² With increasing chloride concentration, the $\text{Bi}^{3+/0}$ redox couple becomes more cathodic, *i.e.* the Bi^{3+} ion is stabilised relative to elemental Bi, meaning that Bi metal should be easier to oxidise. This effect is more significant in the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines, probably because of the higher chloride concentration. The $\text{Sb}^{3+/0}$ couple is found on Sb CVs at around $-0.6 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$ in $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines and around $-0.4 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$ in ChCl brines, while the $\text{Sb}^{5+/3+}$ couple is visible in some CVs for potentials higher than $+0.4 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$. Similarly to the $\text{Bi}^{3+/0}$ couple, the $\text{Sb}^{3+/0}$ couple seems to be more cathodic when the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ content of the brine is increased. However, due to the low solubility of Sb in the least concentrated chloride brines, the lack of data makes the trend difficult to analyse.

The CV of Te in the brines show one couple at around $-0.2 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$ in $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines and around $+0.1 \text{ V vs. } [\text{Fe}(\text{CN})_6]^{3-/4-}$ in ChCl brines which could be attributed to the $\text{Te}^{\text{IV/0}}$ couple, while the second couple appearing on some CVs may be related to the presence of the $\text{Te}^{0/2-}$ couple.⁴⁴ Contrarily to Bi and Sb behaviour, the addition of chloride in the brines causes an anodic shift in the $\text{Te}^{\text{IV/0}}$ couple, meaning that increasing chloride content increases the formal redox potential and that elemental Te becomes harder to oxidise in the brines with the highest chloride content.

The $\text{Sn}^{\text{II/0}}$ and $\text{Cu}^{\text{I/0}}$ couples are visible in ChCl brines at around -0.5 V and -0.4 V , respectively, all *vs.* $[\text{Fe}(\text{CN})_6]^{3-/4-}$. In $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines, the same redox couples are present at

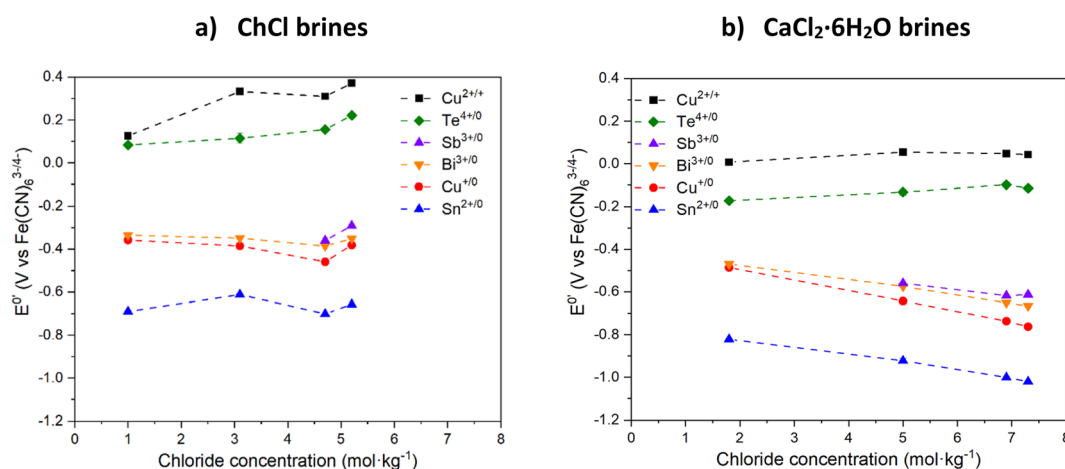


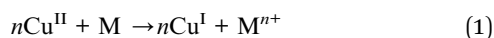
Fig. 1 Formal electrode potentials of the different redox couples in (a) ChCl, and (b) $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines, as a function of solvent chloride content. Values corrected for small changes in concentration and temperature using the Nernst equation, and referenced to the $[\text{Fe}(\text{CN})_6]^{3-/4-}$ couple.



around -0.9 V, and -0.6 V, respectively. For these three couples, increasing the $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ content causes a cathodic shift of the redox potential and makes their oxidation easier. The effect of increasing chloride content is much higher when using $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ as compared to ChCl , probably due to the higher chloride content and a higher impact on speciation.

In all cases, the E_{cell} values for oxidation of the target elements by Cu^{II} are positive, *i.e.* $\Delta G = -nFE_{\text{cell}}$ is negative, indicating that the reaction will be thermodynamically viable regardless of the brine selected. For all the metals investigated except Te, E_{cell} is increasing with increasing $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ content since it causes a slight anodic shift of the redox potential of the $\text{Cu}^{\text{I/0}}$ couple while a cathodic shift is observed for the other metals, except Te. Increasing ChCl content has a much lower effect, with E_{cell} values remaining approximately constant along the series for all elements, except Te.

3.2.2 Leaching of Bi, Cu, Sn, Sb and Te in different solvents. An evaluation of the material recovery of commonly used TEG metals, namely Cu, Sn, Sb, Te and Bi using 0.1 mol dm^{-3} $\text{Cu}(\text{II})$ chloride dissolved in various chloride-containing brines was undertaken. $\text{Cu}(\text{II})$ chloride was selected as an oxidising agent because it minimises contamination of the system with another metal, as would be the case if FeCl_3 was used instead. The oxidation of metals with Cu^{II} is known to occur according to the following equation^{24,45,46} (eqn (1)):



where M represents the metal to be oxidised. However, there are several additional effects that must also be considered, including: (a) Cu^{II} having a much more anodic redox potential compared to $\text{M}^{x+/0}$, and (b) passivation of the target metal *via* the formation of precipitates (whether oxide, hydroxide, or chloride). The corrosivity of the chloride anions towards the metals to be leached was investigated by placing the metals into the different solvents without CuCl_2 present, and it was found that no leaching took place within the timeframe of the experiment.

Metallic Cu is seen to be dissolved in all systems, but the leaching efficiency does not improve as a function of chloride concentration (Fig. 2a and b). The greatest amounts of metal leached were obtained in the solutions containing the highest water contents (*ca.* 0.45 mmol of Cu dissolved over the course of an hour), rather than the highest chloride contents (*ca.* 0.15 to 0.25 mmol of Cu dissolved). In the brines with lower chloride content, white $\text{Cu}(\text{I})$ chloride precipitates were observed that can hinder further leaching. This indicates that the stability of the Cu^{I} species in solution is less important than the increase in solvent viscosity and corresponding decrease in mass transport. The solder metal, Sn, is dissolved at a similar rate to Cu, as it is highly reactive and therefore easily forms stable solution species.

Bi, however, is much more resistant to being leached by Cu^{II} , despite the relative redox potentials indicating that this process is thermodynamically favourable. Fig. 2c and d shows that when the chloride content is less than 3 mol kg^{-1} , no dissolution occurs, and similarly to Cu, a decrease in dissolution rate is seen in the systems with the highest chloride contents, also

being the most viscous liquids. However, even in the system which worked most favourably towards Bi, only *ca.* 0.04 mmol is dissolved – a factor of 10 smaller than the amount for Cu. The oxidation of Sb follows a similar profile to Bi with respect to chloride content, with maximum leaching of *ca.* 0.03 mmol . The leaching of Te with Cu^{II} is poor in all systems investigated here, with a maximum of only *ca.* 0.015 mmol Te oxidised within one hour. Overall, the trend of higher solubility relative to chloride content is maintained, and can be explained by the oxidised species formed; at low chloride content, Bi, Sb and Te ions can form oxide and oxychloride species, which may be passivating. Given that the oxidation of these elements is a 3- or 4-electron process, mass transport is likely to be more critical than for a 1-electron oxidation process. While these poor leaching efficiencies towards the semiconductor elements may appear less than desirable in a recycling process, thermoelectric devices such as the one described in the Experimental section have a layer of Cu that attaches the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ to the ceramic substrates.

Hypothetically, by selectively leaching the Cu away, the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs are released from the ceramic substrate and can be recovered by size grading (as the ceramic substrates are much larger). The solvent will then only contain a mixture of $\text{Cu}(\text{I})$ and $\text{Cu}(\text{II})$ ions, which can then be regenerated to all being $\text{Cu}(\text{II})$ *via* reaction with atmospheric oxygen, or by electrolytic means. The $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs can then be simply rinsed with deionised water to remove any solvent and Cu ions, before further processing. Therefore, an optimal process would ensure maximum Cu and solder leaching efficiency, while minimising solubility of the other elements is the ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines with a chloride concentration of 1.0 and 1.8 mol kg^{-1} , respectively.

3.3 Recovery of $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ from thermoelectric devices

To minimise the number of processing steps required to separate the individual elements from each other, it is advantageous to selectively remove the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs by leaching the Cu tabs that ensure the structural integrity of the thermoelectric materials. It was determined that the most optimal solvent to ensure selectivity towards Cu was the ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines with the lowest chloride content (1.0 and 1.8 mol kg^{-1} , respectively). The two ceramic plates were manually separated from each other to ensure a more efficient removal process, and immersed in the solvents containing 0.1 mol dm^{-3} CuCl_2 as an oxidising agent. Ultrasonication was preferred over mechanical stirring, as it allowed removal of the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs within 30 minutes (with mechanical stirring at 200 rpm and room temperature, it took more than an hour to obtain the same result). After 20 minutes of ultrasonication at room temperature, the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs became separated from the substrate, revealing the underlying Cu layer, partially contaminated with a thin layer of $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ particles (Fig. 3b). The legs remain undissolved and can be easily recovered from the solution *via* filtration, as shown in Fig. 3c, then re-used for the manufacture of new thermoelectric materials, assuming their quality has not been affected by the leaching process. The Cu plates can also be detached from the ceramic substrate and recovered by filtration, as evidenced in Fig. 3c, despite some selective leaching. This could be related to the



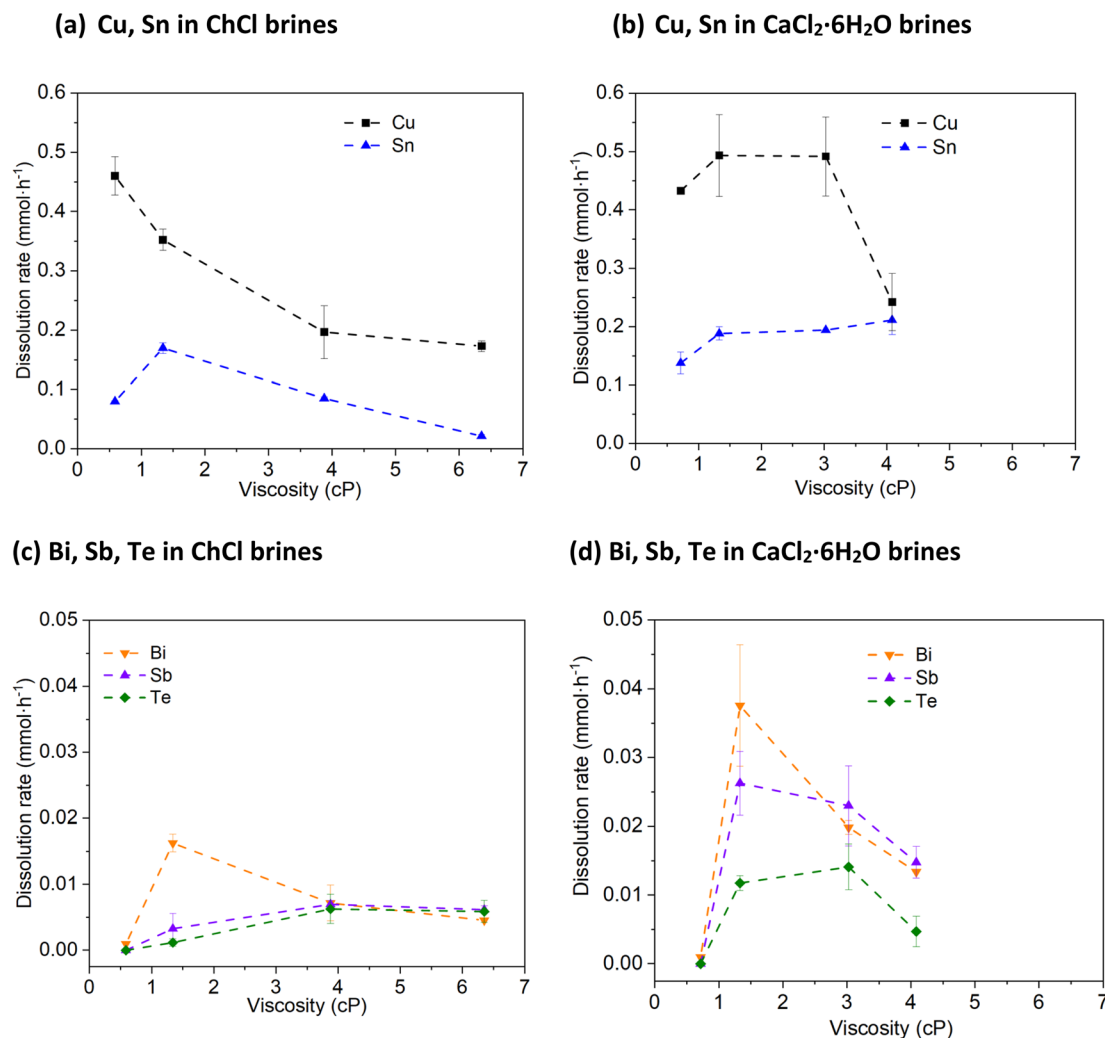


Fig. 2 Rate of metal dissolution using 0.1 mol dm^{-3} of Cu(II) chloride as an oxidising agent for: (a) Cu, Sn in ChCl brines, (b) Cu, Sn in $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines, (c) Bi, Sb, Te in ChCl brines, and (d) Bi, Sb, Te in $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines as a function of solvent viscosity at $50 \text{ }^\circ\text{C}$. Temperature = $50 \text{ }^\circ\text{C}$, stirring rate = 200 rpm .

removal of any solder between the Cu plates and the ceramic substrate, or that the Cu layer has been undercut. Overall, all of the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs are removed from the Cu plate in about 30 minutes. After three repeated contacts with fresh Cu chloride brines, Cu is completely dissolved or removed from the ceramic substrate, as evidenced in Fig. S7.†

Whilst the ultrasonication at room temperature successfully remove the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs, the time taken was 20 min. To increase the separation, alternative approaches were sought to speed up the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ using mechanical separation of both the ceramic substrate and Cu plates without the need for additional chemicals *via* the use of a high intensity ultrasound, as opposed to an ultrasonic bath. As a result of the higher and better directed power density of the ultrasound (198 W cm^{-2}) the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs rapidly broke apart as the sample passed under the tip of the sonotrode. After processing, the solution was filtered to collect the fragmented $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$. The recovered ceramic plate retained a thin layer of $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ attached to the solder on the surface of the Cu tabs (Fig. 4a), indicating that the

recovery percentage of the semiconductor materials (>80%) was less than obtained after chemical leaching (>95%). $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ was recovered as a powder with a wide range of particle sizes (from 1 mm^3 to $10 \text{ }\mu\text{m}^3$), as seen in Fig. 4b. As no material was dissolved during this process, the different products will be easier to separate from one another, while also minimising the amount of waste solvent requiring remediation.

Overall, it is clear that the lower volumes of chemicals, types of chemicals and processing time provides a clear economic benefit over the use of chemical oxidising agents in chloride brines. Further work is clearly needed to establish whether it is preferential to recover material in this manner; whereby TEG material that is recovered in a variety of sizes and needs to be re-processed into thermoelectric legs, or whether the recovery of fully intact thermoelectric legs is a better approach to create a recycled TEG. This would be highly dependent upon the condition and especially, the degradation, of TEG devices when recovered at end-of-life. There are likely to be multiple regulatory and quality control issues in order to directly reuse



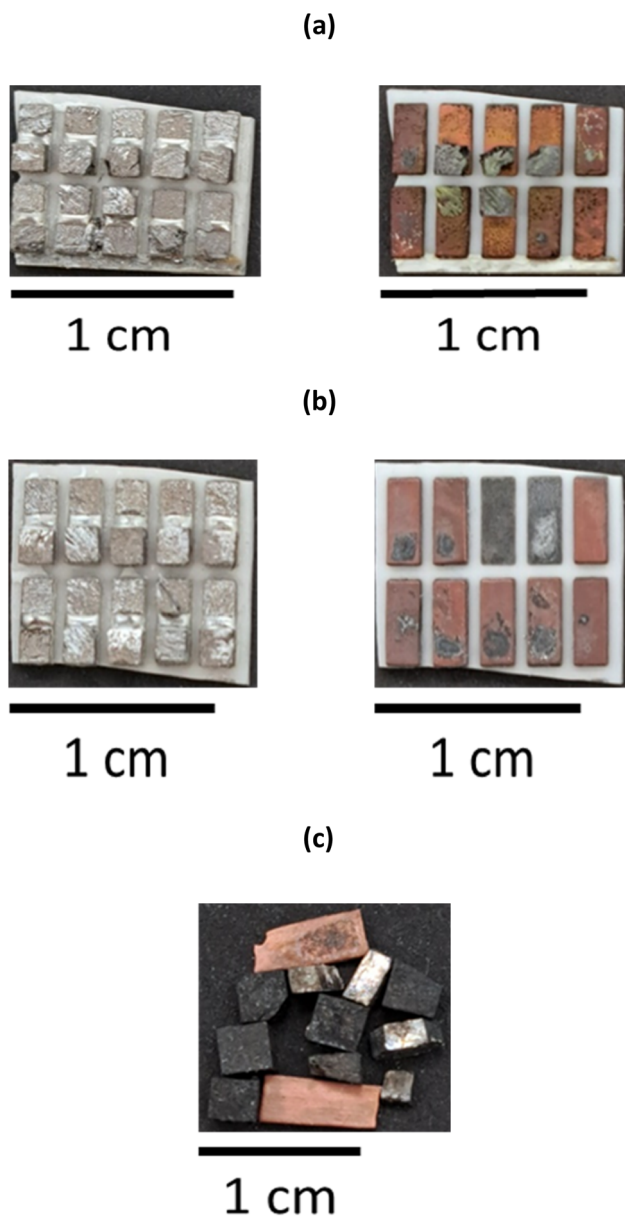


Fig. 3 Photographs of thermoelectrics samples before and after 20 minutes of leaching with 0.1 mol dm^{-3} Cu(II) chloride in (a) ChCl brine (1.0 mol per kg chloride); (b) $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brine (1.8 mol per kg chloride); (c) photograph of the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs and Cu plates recovered from the solution and washed.

thermoelectric legs with both manners. With both techniques, the thermoelectric legs are removed, thus mixing the n-type (which contain Bi, Se and Te) and the p-type (which contain Bi, Sb and Te) legs. Due to the different composition of the legs, further refining is likely needed. Both processes are therefore useful to separate the legs from the initial thermoelectric materials, but require additional steps for either purifying Bi, Sb, Te and Se or remanufacturing p- and n-type thermoelectric legs. High intensity ultrasonication does not involve chemical reactions and is probably more appropriate to avoid chemical modification of the legs (formation of passivation layers...) which could change their properties.

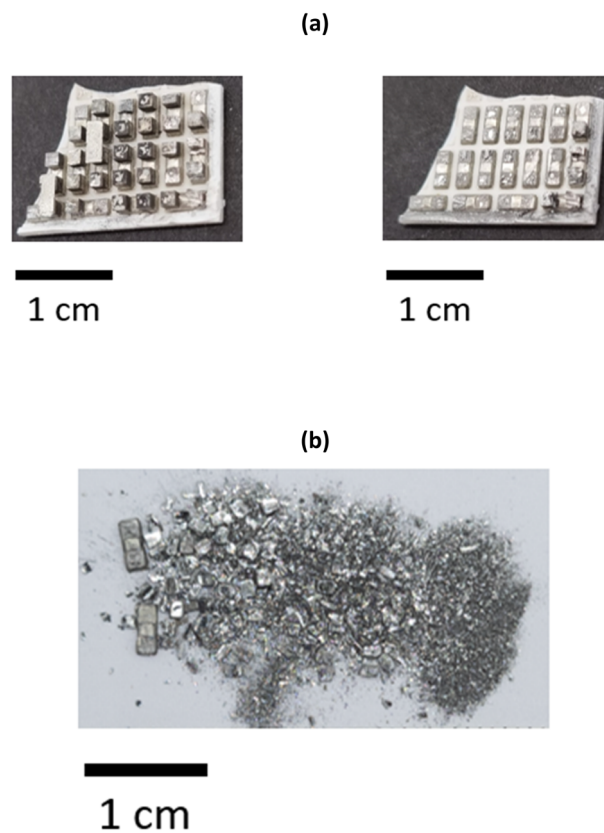


Fig. 4 (a) Photographs of thermoelectric samples before and after 15 seconds of ultrasonication in water. (b) Photograph of the recovered $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ powder.

3.4 Additional concerns

Dissolution of the solder is feasible using different chemicals such as hydrochloric acid (Fig. S8[†]). However, thermoelectric legs are still attached to the copper electrode after removal of the solder. To achieve thermoelectric leg removal, (at least partial) dissolution of the copper electrode is needed. An alternative method to achieve copper dissolution, strong oxidising agents like nitric acid solutions can be used, but it also achieves complete dissolution of the thermoelectric leg, which would require further separation and precipitation of Bi, Sb and Te.

When using copper chloride as a mild oxidising agent, the low amount of nickel (10 mg kg^{-1}), Te ($<2 \text{ mg kg}^{-1}$), Bi ($<1 \text{ mg kg}^{-1}$), selenium ($<1 \text{ mg kg}^{-1}$), Sb ($<0.1 \text{ mg kg}^{-1}$), Sn (*ca.* 160 mg kg^{-1}) and Zn (*ca.* 10 mg kg^{-1}) present in the leaching solution after use makes the recovery of the dissolved elements economically unviable. However, after a currently unknown number of cycles, the leaching solution will become enriched in these elements or precipitation of insoluble species may occur. Recovery methods could include selective electrowinning or precipitation, as seen for Zn in DESs by the addition of ammonia solution.⁴⁷

A potential solution to make the process semi-continuous would be to use a leaching vessel equipped with a sieve, as depicted in Fig. S9[†]. Here, the thermoelectric materials are placed in a leaching solution containing a sieve that has apertures of no more than 2 mm diameter. This sieve will retain the



large ceramic parts (average length 1 cm) while letting the $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs (average size 1 mm^3) pass through, enabling simple gravimetric separation of the components. The used leaching solution could be tapped-off, and regenerated by either using oxygen from air or *via* electrolytic means. After prolonged use of thermoelectric materials, a thin layer of copper oxide could be formed at the surface of the copper electrode, which can partially protect the bulk copper from further oxidation and also reduce the efficiency of the dissolution process. However, copper oxides could be dissolved with acetic acid⁴⁸ or DESs based on organic acids.⁴⁹

Both processes allow the removal of thermoelectric legs without using strong acids and facilitate further processing by separating in one step the legs from the ceramics. Clearly the removal of such acids, with a benign non-acid based method is desirable as it removed the need to handle and dispose of hazardous materials.

With low power ultrasounds, copper oxidation is needed, which requires the use of an oxidising agent, which will require wastewater treatment (although the oxidising agent can be regenerated and the dissolution of elements outside of Cu, Zn and Sn is minimal). In our conditions, the process takes around 20 minutes and requires an electricity consumption of 0.05 kW h per piece of thermoelectric material. With high power ultrasounds, water is used as a propagation media for ultrasound waves. The physical effect of ultrasounds is sufficient to pulverise the brittle thermoelectric legs. This process does not produce wastewater as no elements are dissolved in water. The process is fast (15 seconds) and requires an electricity consumption of around 0.003 kW h. The high-power intensity ultrasonication process looks a more promising processing route considering that it is faster, consumes less electricity and does not produce wastewater. However, it does reduce the thermoelectric legs into pieces of different sizes which may be harder to post-process. Further studies are needed to determine how to remanufacture thermoelectric materials from (relatively) intact pieces or pulverised materials. This would also require further optimisation of the application of ultrasonication.

The two techniques proposed in this manuscript target the weaknesses at the interface between different material layers. The first method allows the dissolution of base metals such as Cu and Sn which have a low redox potential and are relatively easy to oxidise. This technique can potentially be applied to other electronic waste for the selective, acid-free, and low-cost dissolution of soldering elements. It has successfully been demonstrated for the recovery of gold flakes from waste printed circuit boards.²² The second technique takes advantage of the brittle nature of the thermoelectric legs to controllably “blast” the thermoelectric materials with high-power ultrasound. When compared to conventional crushing, it has significant advantages as the approach allows separation of two materials with different brittleness (thermoelectric legs and ceramic) by controllably reducing the size of the most brittle material. This technique is especially important for symbiosis in recycling e-waste when characteristically different brittle materials are present.

4 Conclusions

The electrochemical properties and speciation behaviour of $\text{Cu}(\text{II})$ chloride was investigated in ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines containing varying salt-to-water molar ratios. Increasing chloride content in the brines allows forming anionic chloride complexes which impact the electrochemical behaviour of Cu. Additionally, the speciation and electrochemical of the elements present in thermoelectric materials- (Sb, Bi, Te, Sn) relative to Cu was studied. It was found that chloride species were present for all the elements, except in the most aqueous systems. For all elements to be dissolved, the redox potentials were more cathodic than the oxidising agent, resulting in thermodynamically preferential reactions.

These brines were used for the leaching of pure metals from TEGs, and it was found that in the least concentrated chloride solutions (1.0 mol kg^{-1} and 1.8 mol kg^{-1} chloride for ChCl and $\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$ brines, respectively) showed the greatest selectivity towards Cu and the solder elements. This was determined to be due to passivation of the thermoelectric elements (Bi, Sb, and Te), most likely through the formation of insoluble precipitates. At high chloride contents, the relatively high viscosity of these systems hindered mass transport and decreased leaching rates. This means that for optimal selectivity, there is a balance to be maintained between chloride concentration and solvent properties.

Thermoelectric devices containing $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ were treated with the $1:50\text{ CaCl}_2 \cdot 6\text{H}_2\text{O}$ and ChCl brines. It was found that the Cu and solder elements could be leached away, resulting in pure output streams of clean ceramics and liberated $\text{Bi}_{2-x}\text{Sb}_x\text{Te}_3$ legs which could easily be separated by sieving. When a junction layer of W between the Cu and ceramic was present, the Cu strips were liberated, rather than fully oxidised. The recovery of metallic Cu rather than a dissolved species will improve the viability of the recycling process from an economic point of view, and also result in less change to the leaching bath composition. While the application of high-powered ultrasound to the thermoelectric devices in water resulted in separation of the semiconductor materials from the ceramics and Cu tabs, a powdered material of uneven grain sizes was recovered which would require further processing steps to be useful. Therefore, further work is needed to establish whether fully intact thermoelectric legs are needed for a circular economy for TEGs. Or ultimately TEGs will be treated in the same manner as conventional e-waste, with further size reduction and homogenisation before material recovery.

Conflicts of interest

There are no conflicts of interest to declare.

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References

- 1 C. Forman, I. K. Muritala, R. Pardemann and B. Meyer, *Renewable Sustainable Energy Rev.*, 2016, **57**, 1568–1579.
- 2 M. A. Zoui, S. Bentouba, J. G. Stocholm and M. Bourouis, *Energies*, 2020, **13**, 3606.
- 3 S. B. Riffat and X. Ma, *Appl. Therm. Eng.*, 2003, **23**, 913–935.
- 4 A. Bahrami, G. Schierning and K. Nielsch, *Adv. Energy Mater.*, 2020, **10**, 1904159.
- 5 N. T. Nassar, H. Kim, M. Frenzel, M. S. Moats and S. M. Hayes, *Resour., Conserv. Recycl.*, 2022, **184**, 106434.
- 6 E. Rombach and B. Friedrich, *Handbook of Recycling*, Elsevier, 2014, pp. 125–150.
- 7 M.-A. Shahbazi, L. Faghfour, M. P. A. Ferreira, P. Figueiredo, H. Maleki, F. Sefat, J. Hirvonen and H. A. Santos, *Chem. Soc. Rev.*, 2020, **49**, 1253–1321.
- 8 A. Chatterjee, C. N. Lobato, H. Zhang, A. Bergne, V. Esposito, S. Yun, A. R. Insinga, D. V. Christensen, C. Imbaquingo, R. Bjørk, H. Ahmed, M. Ahmad, C. Y. Ho, M. Madsen, J. Chen, P. Norby, F. M. Chiabrera, F. Gunkel, Z. Ouyang and N. Pryds, *JPhys Energy*, 2023, **5**, 022001.
- 9 N. Jaziri, A. Boughamouira, J. Müller, B. Mezghani, F. Tounsi and M. Ismail, *Energy Rep.*, 2020, **6**, 264–287.
- 10 G. Calvo, G. Mudd, A. Valero and A. Valero, *Resources*, 2016, **5**, 36.
- 11 N. Rötzer and M. Schmidt, *Resources*, 2018, **7**, 88.
- 12 D. Champier, *Energy Convers. Manage.*, 2017, **140**, 167–181.
- 13 *Selenium and Tellurium Chemistry: from Small Molecules to Biomolecules and Materials*, ed. J. D. Woollins and R. S. Laitinen, Springer, Heidelberg; New York, 2011.
- 14 E. Deady, C. Moon, K. Moore, K. M. Goodenough and R. K. Shail, *Ore Geol. Rev.*, 2022, **143**, 104722.
- 15 V. A. Krenev, N. F. Drobot and S. V. Fomichev, *Theor. Found. Chem. Eng.*, 2015, **49**, 540–544.
- 16 O. Velázquez-Martinez, A. Kontomichalou, A. Santasalo-Aarnio, M. Reuter, A. J. Karttunen, M. Karppinen and R. Serna-Guerrero, *Resour., Conserv. Recycl.*, 2020, **159**, 104843.
- 17 W.-B. Kim, *Bull. Korean Chem. Soc.*, 2013, **34**, 2167–2170.
- 18 H. Y. Lee and J. K. Lee, *Sep. Sci. Technol.*, 2015, **50**, 1665–1670.
- 19 R. Sasai, T. Fujimura and T. Sano, *J. Ceram. Soc. Jpn.*, 2021, **129**, 118–121.
- 20 W. D. Bonificio and D. R. Clarke, *J. Appl. Microbiol.*, 2014, **117**, 1293–1304.
- 21 G. R. T. Jenkin, A. Z. M. Al-Bassam, R. C. Harris, A. P. Abbott, D. J. Smith, D. A. Holwell, R. J. Chapman and C. J. Stanley, *Miner. Eng.*, 2016, **87**, 18–24.
- 22 R. Marin Rivera, G. Zante, J. Hartley, K. S. Ryder and A. P. Abbott, *Green Chem.*, 2022, **24**, 3023–3034.
- 23 D. L. Thompson, I. M. Pateli, C. Lei, A. Jarvis, A. P. Abbott and J. M. Hartley, *Green Chem.*, 2022, **24**, 4877–4886.
- 24 G. Zante, R. Marin Rivera, J. M. Hartley and A. P. Abbott, *J. Cleaner Prod.*, 2022, **370**, 133552.
- 25 M. F. C. Carneiro and V. A. Leão, *Hydrometallurgy*, 2007, **87**, 73–82.
- 26 J. E. Dutrizac, *Hydrometallurgy*, 1992, **29**, 1–45.
- 27 P. R. Holmes and F. K. Crundwell, *Geochim. Cosmochim. Acta*, 2000, **64**, 263–274.
- 28 J. Liddicoat and D. Dreisinger, *Hydrometallurgy*, 2007, **89**, 323–331.
- 29 G. Steinhauser, *J. Cleaner Prod.*, 2008, **16**, 833–841.
- 30 C. D'Agostino, R. C. Harris, A. P. Abbott, L. F. Gladden and M. D. Mantle, *Phys. Chem. Chem. Phys.*, 2011, **13**, 21383.
- 31 J. M. Hartley, J. Allen, J. Meierl, A. Schmidt, I. Krossing and A. P. Abbott, *Electrochim. Acta*, 2022, **402**, 139560.
- 32 F. Bevan, H. Galeb, A. Black, I. M. Pateli, J. Allen, M. Perez, J. Feldmann, R. Harris, G. Jenkin, A. Abbott and J. Hartley, *ACS Sustainable Chem. Eng.*, 2021, **9**, 2929–2936.
- 33 C. Merritt, H. M. Hershenson and L. B. Rogers, *Anal. Chem.*, 1953, **25**, 572–577.
- 34 H. Nikol and A. Vogler, *J. Am. Chem. Soc.*, 1991, **113**, 8988–8990.
- 35 L. Vieira, J. Burt, P. W. Richardson, D. Schloffer, D. Fuchs, A. Moser, P. N. Bartlett, G. Reid and B. Gollas, *ChemistryOpen*, 2017, **6**, 393–401.
- 36 B. E. Etschmann, W. Liu, A. Pring, P. V. Grundler, B. Tooth, S. Borg, D. Testemale, D. Brewe and J. Brugger, *Chem. Geol.*, 2016, **425**, 37–51.
- 37 G. S. Pokrovski, A. Yu. Borisova, J. Roux, J.-L. Hazemann, A. Petdang, M. Tella and D. Testemale, *Geochim. Cosmochim. Acta*, 2006, **70**, 4196–4214.
- 38 M. Filella, N. Belzile and Y.-W. Chen, *Earth-Sci. Rev.*, 2002, **59**, 265–285.
- 39 D. M. Sherman, K. V. Ragnarsdottir, E. H. Oelkers and C. R. Collins, *Chem. Geol.*, 2000, **167**, 169–176.
- 40 J. M. Hartley, C.-M. Ip, G. C. H. Forrest, K. Singh, S. J. Gurman, K. S. Ryder, A. P. Abbott and G. Frisch, *Inorg. Chem.*, 2014, **53**, 6280–6288.
- 41 N. Shekhovtsova, T. Glyzina, S. Romanenko and N. Kolpakova, *J. Solid State Electrochem.*, 2012, **16**, 2419–2423.
- 42 L.-Y. Hsieh, J.-D. Fong, Y.-Y. Hsieh, S.-P. Wang and I.-W. Sun, *J. Electrochem. Soc.*, 2018, **165**, D331–D338.
- 43 J. A. Barragan, C. Ponce de León, J. R. Alemán Castro, A. Peregrina-Lucano, F. Gómez-Zamudio and E. R. Larios-Durán, *ACS Omega*, 2020, **5**, 12355–12363.
- 44 A. Sorgho, M. Bougouma, G. De Leener, J. Vander Steen and T. Doneux, *Electrochem. Commun.*, 2022, **140**, 107327.
- 45 O. Cakir, *J. Mater. Process. Technol.*, 2006, **175**, 63–68.
- 46 T. Keskitalo, J. Tanskanen and T. Kuokkanen, *Resour., Conserv. Recycl.*, 2007, **49**, 217–243.
- 47 A. P. Abbott, J. Collins, I. Dalrymple, R. C. Harris, R. Mistry, F. Qiu, J. Scheirer and W. R. Wise, *Aust. J. Chem.*, 2009, **62**, 341.
- 48 K. L. Chavez and D. W. Hess, *J. Electrochem. Soc.*, 2001, **148**, G640.
- 49 A. P. Abbott, G. Capper, D. L. Davies, K. J. McKenzie and S. U. Obi, *J. Chem. Eng. Data*, 2006, **51**, 1280–1282.

