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A self-powered photodetector through facile processing using polyethyleneimine/carbon quantum dots for highly sensitive UVC detection†

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Ultraviolet C (UVC) photodetectors have garnered considerable attention recently because the detection of UVC is critical for preventing skin damage in humans, monitoring environmental conditions, detecting power aging in facilities, and military applications. As UVC detectors are “solar-blind”, they encounter less interference than other environmental signals, resulting in low disturbance levels. This study employed a natural precursor (glucose) and a one-step ultrasonic reaction procedure to prepare carbon quantum dots (CQDs), which served as a convenient and environmentally friendly material to combine with polyethyleneimine (PEI). The prepared materials were used to develop a self-powered, high-performance UVC photodetector. The thickness of the constitutive film was investigated in detail based on the conditions of the electron transport pathway and trap positions to further improve the performance of the PEI/CQD photodetectors. Under the optimized conditions, the photodetector could generate a strong signal (1.5 mA W^{-1} at 254 nm) and exhibit high detectability (1.8×10^{10} Jones at 254 nm), an ultrafast response, and long-term stability during the power supply sequence. The developed solar-blind UVC photodetector can be applied in various ways to monitor UVC in an affordable, straightforward, and precise manner.

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1. Introduction

Although the ultraviolet (UV) region comprises 10% of the total solar radiation, it has an immense effect on ecosystems, including human life. UV light is extensively used in many fields, including polymer curing in semiconductor fabrication, lithography, subcutaneous exploration in the medical field, germ sterilization, air disinfection, and flare detection in military applications.^{1,2} The detection of UV C-band (UVC) or deep-UV emissions can potentially be used as a predictive warning signal for corona discharge or flame detection, which is essential for human safety. UVC exposure is extremely high for arc welders and scientists who operate with open arc and specialized arc lamps. There has been considerable interest in monitoring the UVC levels in these specialized areas. Furthermore, increasing UVC radiation may affect residential areas because of ozone layer depletion. It is generally known that even at extremely low exposure levels, UVC can have a significantly negative impact, which is typically manifested as acute and chronic dermatitis.^{3–6} For instance, Trevisan *et al.* reported that acute exposure to short wavelengths (245–290 nm) can damage mammalian cells at the DNA level.⁷ Abnormal skin reactions

have been observed in welders.⁸ Acute sunburn was reported when using electricity to kill insects with equipment that emitted up to 46 mW m^{-2} of radiation.⁹ Nevertheless, it is difficult to perceive low-intensity UVC light in surroundings lit with ordinary visible light (sunlight). Thus, it is critical to develop a sensitive photodetector (PD) to monitor UVC radiation during daylight hours.^{10–12}

In recent years, wide bandgap semiconductors ($E_g > 4.43 \text{ eV}$), such as AlGaIn, MgZnO, $\beta\text{-Ga}_2\text{O}_3$, GaN, NiO, BN, and LaAlO₃, have been increasingly used to detect solar-blind UV. Studies related to UVC detectors and their respective performance are presented in Table 1. These materials have demonstrated high sensitivity,^{13–15} flexible detection,¹⁶ and operation in severe conditions.¹⁷ Unfortunately, the production techniques for these PDs are intricate and expensive, and the lattice mismatch between the film and the substrate is often a concern for their applications.¹⁸ Besides, the bandgap of a semiconductor must surpass 4.4 eV to detect UVC light, which greatly limits the choices of semiconductor materials.¹⁹ Therefore, finding suitable materials and simple manufacturing techniques are critical challenges for fabricating high-performance PDs. Organic photodetectors (OPDs) have garnered significant attention for the upcoming era of flexible electronics owing to their potential as flexible and lightweight PDs. OPDs have high photodetection capabilities and can be easily tuned to achieve specific optoelectronic features, such as charge production, transport, and recombination, by adjusting their molecular architectures.²⁰

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Table 1 Performance parameters of different UVC PD devices

Photo-absorbing materials	Voltage (V)	Wavelength (nm)	Responsivity (mA W ⁻¹)	Detectivity (Jones)	Thickness (nm)	Ref.
AlGaIn	20	269	2.34×10^3	—	600	64
MgZnO	—	254	0.22	4.00×10^{10}	20	65
Ga ₂ O ₃	5	260	2.4×10^{10}	1.70×10^{15}	270	66
ZnGa ₂ O ₄	5	233	86.3×10^3	2.45×10^8	—	67
BN	20	212	0.1	2.4×10^8	3	68
LaAlO ₃	10	210	71.8	—	500	69
In ₂ Ge ₂ O ₇	3.0	290	3.9×10^8	—	30	70
CuO	0.2	280	1.03×10^4	—	45	71
CsPbBr ₃	-0.1	279	1.4	2.4×10^{11}	10^3	72
ZnS QD	15	254	1.6	0.55×10^{10}	—	73
ZnS QD	40	265	0.1	—	130	74
NiO	0.05	250	—	2.44×10^{11}	150	75
ZnS QD	Self-power	250	0.29	1.41×10^{10}	315	18
GaN	Self-power	244	14.9	8.1×10^{11}	3×10^3	76
SnO ₂ QD	Self-power	254	49.6	2.16×10^{10}	111	19

Another advantage of OPDs is that their detection band (wavelength range) can be easily controlled by altering the organic active materials. Moreover, OPDs are solution-processable, and the printable electroactive inks enable large-scale, low-cost manufacture on flexible substrates.^{21–23} More significantly, human tissue has an approximate average density of 1.10 g cm^{-3} , with soft tissues frequently targeted in medical dosimetry having a density of 1.07 g cm^{-3} . Thus, the density of thin solid films made from organic electroactive inks is very similar to that of human tissue.²⁴ Because of these benefits, organic electronic materials have become increasingly popular in a wide range of applications, such as thin-film transistors,²⁵ solar cells,^{26–28} sensors,²⁹ visual displays,³⁰ PDs,³¹ and the medical devices.³² According to a recent study on polymer-based UV PDs, organic semiconductors produce good UV PDs owing to their selectivity and sensitivity at the wavelengths required for diverse applications.^{33–38} The advantages of all-organic UV PDs include straightforward processing methods, inexpensive material costs, functional tunability, and the ability to process on flexible substrates.^{39–42} The chemical and physical features of organic materials can be modified to reinforce their bandgap and adjust their chemical composition for more UV absorption. Selecting materials with excellent absorption properties and an internal network of donors and acceptors can effectively convert photons to electrons, boost the mobility of the high-charge carriers,⁴³ and create more stable charge carriers among the thin solid films. Active organic layers with bulk heterojunctions formed by blending hole-transporting and electron-transporting materials are typically utilized in device manufacturing. Conductive polymers have found extensive applications in UV PDs, supercapacitors, batteries, and electrochemical detection owing to their high transmittance in the UV-vis region, excellent hole transport capacity, straightforward processing, conjugated π -electrons, excellent flexibility, favorable stability, amenability to deposition by solution-processing, and low material cost. Consequently, conjugated polymers are recognized as prospective candidate materials for combining n-type semiconductors and p-n heterojunction solar devices.⁴⁴

Among different conductive polymers, polyethyleneimine (PEI) is an inexpensive and accessible polymer that can modify the nucleation of ZnO nanorods on a polycrystalline ZnO seed layer, thus modifying the aspect ratio of ZnO nanorods. PEI can be adsorbed on ZnO in an orientation with the m-planes parallel to the surface, which slows the nucleation rate of ZnO and causes a sizable vertical misalignment.⁴⁵ Ultralong ZnO nanorods can be achieved by adding PEI to the solution used for standard hydrothermal synthesis. Qiu *et al.* generated $\sim 40 \mu\text{m}$ long, vertically aligned ZnO nanorod arrays after numerous growth cycles using a PEI-assisted preheating hydrothermal technique.⁴⁶ Burke-Govey *et al.* controlled the diameter and length of ZnO nanowires using PEI with distinct molecular weights at various concentrations.⁴⁷ Additionally, Chen *et al.* attached PEI to modify the structural and optical characteristics of ZnO nanowires and established a connection between the visible emissions and structural defects in ZnO nanowires.⁴⁸ Zhou *et al.* were the first to employ aliphatic amine-rich polymers like PEI as surface modifiers for modulating electrical interactions.⁴⁹ The presence of PEI improved the power detection boundary and specific detection ability by lowering the dark current of the device. Cesarini *et al.* used non-toxic solvents instead of 2-methoxyethanol to print PEI on different substrates,⁵⁰ and the effects of different parameters used for depositing PEI interlayer on the repeatability of the performance of the inkjet-printed devices were studied. The use of multiple solvents significantly increased the yield (from less than 20% to more than 90%). Furthermore, the dark current was lowered to 57 nA cm^{-2} . Because of the interface between the PEI interlayer and the photoactive layer, these devices also had faster response times of up to 30 s. Despite the numerous advantages of PEI-based devices, the combination of PEI and organic compounds has rarely been investigated.

Carbon quantum dots (CQDs) have received widespread attention from the scientific community owing to their promising photoluminescence and electronic properties. CQDs have found wide applications in light-emitting diodes,⁵¹ chemical sensors,⁵² photocatalysts,^{53–55} fluorescent probes,^{56,57} optical sensitizers,⁵⁸ photovoltaics,⁵⁹ and DNA detection.⁶⁰ In contrast to compound semiconductors, such as perovskites, silicon, germanium, and



gallium arsenide, all forms of carbon-based materials have unique environmental benefits, including earth-abundance, low toxicity, chemical stability, and biocompatibility. CQDs have been investigated as photosensitive materials for photodetection applications in several anchored or hybrid architectures. The zero-dimensional confinement exhibited by quantum dots facilitates inter-sub-band transitions between quantized energy levels, thereby enabling broadband photodetection. Lee *et al.* explored a ZnO nanorod/CQD composite as a feasible PD and found that CQDs could act as an electron–hole separator when exposed to UV light.⁶¹ Similarly, Li *et al.* employed CQDs to separate the excitons produced by photosynthesis in TiO₂ to reduce the dark current and increase the reaction time.⁶² Furthermore, Sahatiya *et al.* created a hybrid structure using CQDs and MoS₂ on a glass substrate, which could be used for broadband photodetection from the UV to near-infrared (NIR) region. CQDs demonstrate the advantages of high mobility of photogenerated carriers, tunable bandgaps, ability to function as both an electron donor and an acceptor, high molar extinction coefficients and light absorption coefficients, and wide absorption wavelength ranges. However, CQD-based PDs have been scarcely reported because of their low responsiveness and slow reaction speed.⁶³

In this study, a UVC PD with an intrinsically wide bandgap is fabricated using CQDs and PEI. A straightforward spin-coating process is used to prepare the PD with an asymmetric pair of Schottky contacts, including PEI-coated fluorine-doped tin oxide (FTO)/CQD and poly(3,4-ethylenedioxythiophene):polystyrene sulfonate (PEDOT:PSS)/CQD. The as-fabricated device demonstrates a satisfactory response to 254 nm UV radiation with a self-powered function. The CQD-based PD also showed a high response (1.5 mA W⁻¹ at 254 nm), high detection ability (1.8 × 10¹⁰ Jones at 254 nm), and an ultrafast response time. Moreover, after 50 cycles, the photo-response of the unencapsulated UVC PD remains consistent. The CQDs improve the conductivity of the composite for carrier transport by reducing the interfacial trap states in the heterojunction. The recommended device design utilizes CQDs in conjunction with PEI. These components can be readily combined on an FTO substrate using simple fabrication methods that are appropriate for large-scale manufacturing.

2. Experimental

2.1. Materials

D-Glucose (99.5%), hydrochloric acid (37%), sodium hydroxide (97%), magnesium sulfate heptahydrate (≥99%), polyethylenimine (PEI; branched, *M_w* = 25 000), ethyl alcohol (anhydrous, 99.9%), 2-propanol (IPA; 99.5%), *N,N*-dimethylformamide (DMF; anhydrous, 99.8%), dimethyl sulfoxide (DMSO; anhydrous, ≥99.9%), and poly(3,4-ethylenedioxythiophene)–poly(styrenesulfonate) (PEDOT:PSS; Orgacon dry redispersible pellets) were purchased from Sigma-Aldrich.

2.2. Synthesis of CQDs

A single-step, alkali-assisted, ultrasonic chemical process was used to prepare CQDs.⁷⁷ In a typical synthesis, 9.0 g of glucose

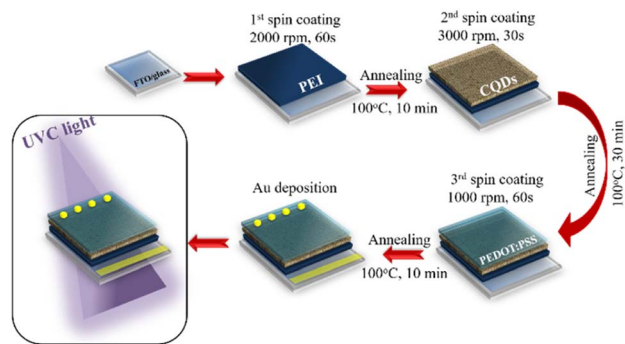


Fig. 1 Schematic of fabrication of a UVC PD.

was placed in a glass beaker, and 50 mL of deionized water was added. Subsequently, the glucose solution was subjected to 4 h of room-temperature ultrasonic treatment (300 W, 40 kHz) with an equal volume of NaOH solution (1 M). A specific amount of ethanol was gradually added to the unreacted solution of glucose and NaOH with constant stirring. HCl was added to the solution to increase the pH to 7. The solution was then treated by adding a suitable amount of MgSO₄ (10–12% weight percent) with stirring for 20 min, followed by storage for 24 h to remove the salts and water.

2.3. Fabrication of PD

A schematic of the PD fabrication is shown in Fig. 1. A UVC PD was constructed in a vertical configuration using a straightforward solution process. The FTO-coated glass substrate was cleaned in an ultrasonic bath for 10 min using acetone, deionized water, and ethanol. After spin-coating an interfacial layer of PEI in IPA (solution concentration: 1, 3, 5, 7, and 9 mg mL⁻¹) onto the cleaned substrate for 60 s at 2000 rpm, the layer was heated for 10 min at 100 °C on a hot-plate. By spin-coating the CQD suspension (dilute concentration in ethanol solution: 20, 30, 40, 50, and 60 wt%) at 3000 rpm for 30 s and annealing at 100 °C for 10 min, the UVC absorber layer was created. PEDOT:PSS (1.3 wt% in IPA with 5 wt% supporting DMSO) was spin-coated on the surface of the CQDs for 60 s at 1000 rpm and heated at 100 °C for 10 min. Ultimately, the outermost layer was deposited by thermally evaporating 100 nm of Au on top of the device under high vacuum pressure (8 × 10⁻⁶ Torr).

2.4. Characterization and measurement

The morphological and crystallographic characteristics of the as-synthesized CQDs were analyzed using scanning electron microscopy (SEM; Hitachi SU8600, Tokyo, Japan) and X-ray diffraction (XRD; SmartLab, Rigaku, Tokyo, Japan). In addition, UV-vis spectrophotometry (LAMDA750, PerkinElmer, Waltham, USA) was used to investigate the optical properties and bandgaps of the CQDs. Fourier-transform infrared spectroscopy (FT-IR) (Thermo Fisher Scientific (iS50)) was used to examine the surface functional groups of the CQDs. The photoluminescence (PL; Fliorolog-QM, Horiba, Japan) of the CQDs was measured at an excitation wavelength of 260 nm. Cross-



sectional SEM was used to measure the thickness and surface roughness of the CQD and PEI films. To evaluate the electrical characteristics of the fabricated PD, current–time ($I-t$) and current–voltage ($I-V$) profiles were recorded using a source meter (Keithley 2400, Tektronix, Beaverton, OR, USA) under UV irradiation at 254 and 365 nm (Spectroline UV-5NF, Spectronics Corporation, Melville, NY, USA). The potential was recorded using a potentiostat–galvanostat (Bio-Logic SP-240). A 254 nm UV lamp (Vilber Lourmat VL-6.LC) was used to illuminate the PD to measure the photo-response. Simulated solar irradiation was provided by a solar simulator (Polaromix K201, Solar simulator LAB 50, McScience K3000) with an irradiance of 100 mW cm⁻². Electrical signals were recorded using a Keithley source meter (2400).

3. Results and discussion

3.1. Characterization of photoactive materials

CQDs were prepared using a natural precursor (glucose) and a one-step ultrasonic reaction procedure, as shown in Fig. 2a. These CQDs served as a convenient and environmentally friendly platform for producing PD devices.⁷⁷

The emission properties and quality of the CQDs remained stable after storage in air for several months at ambient temperature, confirming their excellent optical stability. These CQDs could be easily dispersed in water without ultrasonication, facilitating their widespread application in PD devices.³⁷ Furthermore, the advantages of CQDs, such as excellent optical properties, high response, and high chemical

stability, will be demonstrated in this study. From the SEM image (Fig. 2b), the CQDs are aggregates of nanoparticles with a size of 2–9 nm, dominated by 5 nm particles. The peak at 25.9° in the XRD pattern, corresponding to the (002) plane, demonstrates the successful synthesis of the CQDs (Fig. 2c).⁷⁸ Concurrently, FT-IR analyses were conducted to determine the surface functional groups on the CQDs. The distinctive O–H absorption band (3413 cm⁻¹), C=O stretching vibration band (1651 cm⁻¹), and C–O stretching vibration band (1023 cm⁻¹) demonstrate the presence of the carboxylic group.⁷⁹ Additionally, the weak signal at 2295 cm⁻¹, attributed to –C≡N, indicates the presence of amino-containing functional groups, which are required for streptavidin conjugation.^{80,81} The optical properties of the CQDs are displayed in Fig. 2e. The CQDs could absorb light in UVC range selectively, whereas the absorbance in the UVB range was minimal (visible blindness). Owing to the quantum confinement effect, the optical bandgap estimated from the Tauc plot is 4.1 eV (as shown in Fig. 2f), which is significantly greater than that of other carbon-based materials. Because of their large bandgap and selective absorption ability in UVC range, the prepared CQDs are excellent materials for UVC PDs. The transmittance of the CQDs in the UVC range was significantly low (~4% at 254 nm; Fig. S1a†). Nevertheless, because CQDs are essential materials with high electron mobility, they cannot absorb light in the UVC region (Fig. S1b†). The functional groups on the surface of the CQDs formed an insulating shell, and the repulsive forces between the functional groups assisted in dispersing the CQDs, which retarded electrical transport, generated an inevitable leakage current, and reduced light absorption in the UVC region. Decreasing the concentration of CQDs can reduce the number of similarly charged particles (repulsive effect) and improve the charge transport pathway in the film, thus promoting the detection of UVC light. The CQD solutions were diluted to 60, 50, 40, 30, and 20 wt%, and the corresponding samples are denoted as CQDs-60, CQDs-50, CQDs-40, CQDs-30, and CQDs-20, respectively. The PL spectra (Fig. 2g) of the CQDs displayed two emission peaks at approximately 259 and 490 nm following excitation at 260 nm. The emission of the intermediate states of the CQDs resulting from surface defects, such as oxygen vacancies, is indicated by the peak at approximately 490 nm, whereas the peak at 259 nm represents the recombination of excitations and return to the ground state.

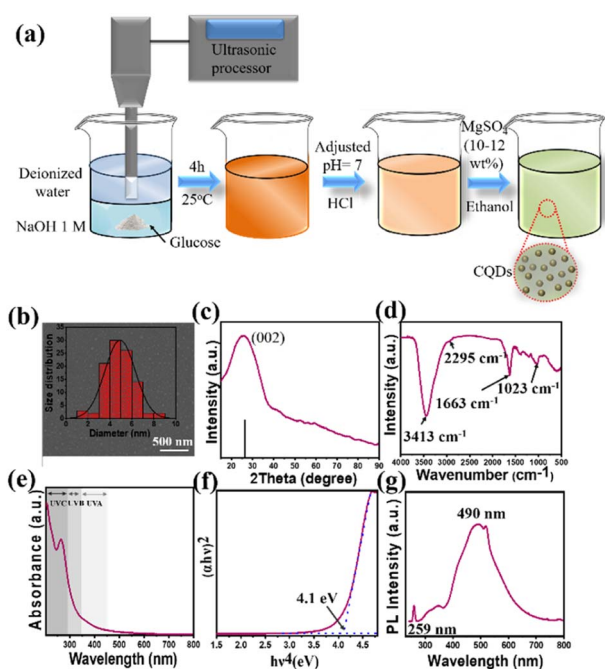


Fig. 2 (a) Schematic of synthesis of CQDs, (b) SEM image of CQDs coated on glass (inset: size distribution of CQDs), (c) XRD pattern of CQDs, (d) FT-IR spectrum of CQDs, (e) UV-vis absorbance spectrum of CQDs, (f) Tauc plot of CQDs, and (g) PL spectrum (excitation wavelength of 260 nm) of CQDs.

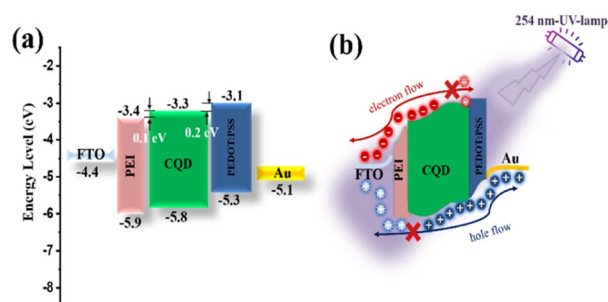


Fig. 3 (a) Energy diagram of CQD-based PD device, and (b) operating mechanism of UVC detector under 254 nm illumination.



To understand the photoelectric conversion mechanism of the prepared PD, Fig. 3a shows the energy band diagrams of individual constituents of the device, including their conduction bands (CB) and valence bands (VB). FTO and Au were used as the bottom and top electrodes for the collection of electrons and holes, respectively.^{82–85} Here, a thin film of PEI is sandwiched between the CQD and FTO/glass substrate. The PEI interfacial layer supports close film formation and separates the top and bottom electrodes. The VB of the PEI layer was lower than that of the CQDs, which hindered hole transport from the CQD to the FTO. Furthermore, the CB of PEI is located between that of CQD and the Fermi level of FTO, facilitating electron extraction from CQD to FTO. The PD can operate without an external bias voltage owing to the difference in the Schottky barrier height between the PEI-coated FTO/CQD and PEDOT:PSS/CQD contacts. It is observed that the energy barriers at PEI-coated FTO/CQD and PEDOT:PSS/CQD interfaces are approximately 0.1 and 0.2 eV, respectively. Consequently, these energy barriers prevent charge carriers from migrating between electrodes, which is a main contributor to dark current. When the device is exposed to 254 nm UV illumination, electron–hole pairs are generated because the energy of the photons is greater than the bandgap of the CQD (as shown in Fig. 3b). The photoexcited electrons can overpass the lower Schottky barrier (0.1 eV) and be collected at the FTO side. For charge neutralization, the photogenerated holes concurrently move to the PEDOT:PSS side, generating a photocurrent at zero bias.

3.2. Surface morphology of PEI and CQD films

The uniformity of the film significantly affects the performance of the PDs; for instance, an uneven surface can reduce energy loss by mitigating the scattering and reflection of incident light

and improving trapping between the interfaces. Poor film uniformity also reduces the bonding between the interface layers, leading to a higher contact resistance. The morphologies of the spin-coated PEI (7 mg mL⁻¹) and CQDs (CQDs-30) on the FTO/glass substrates are displayed in Fig. 4a and b. Both films were uniform, continuous, and devoid of any obvious cracks, flaws, or voids and could be confidently applied to create vertically stacked devices. The thickness of the film at different concentrations of PEI and CQD was investigated because this parameter significantly affects the ability of the device to absorb light (as shown in Fig. 4c, d, S2 and S3†). The PEI film was very thin (<100 nm in different solution concentrations), and its thickness was expected to positively affect light detection owing to a significant reduction in the number of trap positions and light penetration loss. An excessively thick CQD layer (>500 nm) increased the number of electron transport channels, creating favorable conditions for light penetration. Thus, a synergy between the PEI and CQD layers contributed to improving the device performance. Furthermore, the film thickness varied linearly with the concentration of both PEI and the CQDs. Increasing the PEI concentration from 1 to 9 mg mL⁻¹ changed the film thickness from 51 to 86 nm (Fig. S2†). By contrast, when the concentration of the CQDs was decreased from 60 to 20 wt%, the film thickness gradually decreased from 782 to 701 nm because of the smaller number of CQDs in the ethanol solution (Fig. S3†). The thicknesses of the PEI and CQD films with respect to the solution concentration and weight percentage are shown in Fig. 4e and f. As the concentration of the solution increased, the thickness of the PEI film increased linearly (coefficient of determination (R^2) = 0.9995). Moreover, in the case of the CQDs, decreasing the weight percentage of CQDs in the solution linearly reduced the film thickness (R^2 = 0.9993). The high chemical stability of the CQDs was demonstrated *via* contact angle analysis (Fig. S4†). As the PEI concentration increased from 1 to 9 mg mL⁻¹, the contact angle increased from 23.3° to 30.3°. Nevertheless, when the CQD concentration was varied from 60 to 20 wt%, the contact angle was stabilized at approximately 26°. The excellent chemical stability of the CQDs is favorable for large-scale applications.

3.3. Effect of layer thickness on PD performance

We first investigated the influence of the PEI layer on the PD performance of the FTO/glass/CQDs/PEDOT:PSS/Au device. The responsivity (R) of the PD was calculated as follows:

$$R = \frac{I_{\text{photo}} - I_{\text{dark}}}{PA} \quad (\text{A W}^{-1})$$

where P is the light intensity, A is the active area of the PD, and I_{dark} and I_{photo} are the currents under dark and illuminated conditions, respectively.

As shown in Fig. S5a and b,† the dark current level of the device with PEI was higher than that without PEI at zero bias. However, when illuminated at 254 nm, the photocurrent generated by the device with PEI was significantly higher (86 nA) than that without PEI (39 nA) (Fig. S5c†). Because of the decreased photocurrent, the device with PEI exhibited a higher responsivity than that without PEI. Fig. S6 and S7† show the I – V

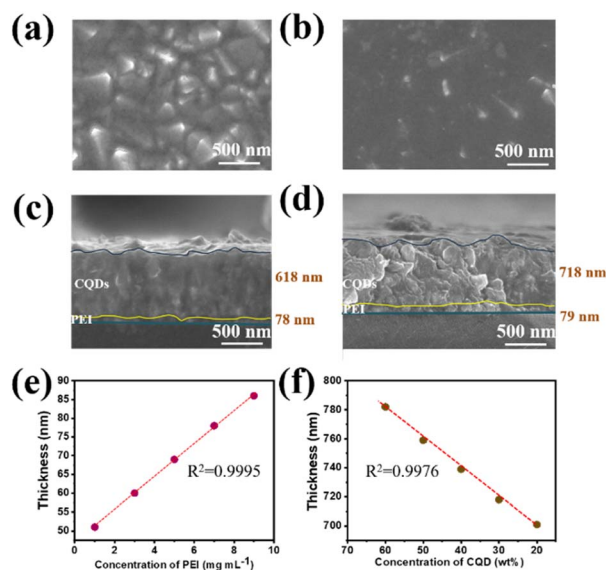


Fig. 4 SEM image of (a) PEI layer (5 mg mL⁻¹), (b) CQD layer (50 wt%). Cross-sectional SEM image of (c) PEI (5 mg mL⁻¹), (d) CQDs (50 wt%). The film thickness changed as a function of the concentrations of (e) PEI and (f) CQDs in the solution.



profiles of the PEI and CQD single-layer devices (FTO-glass/PEI/Au and FTO-glass/CQDs/Au). A linear I - V profile was detected, suggesting ohmic contacts were formed at the semiconductor/electrode interfaces because PEI had a higher Fermi level than both electrodes (FTO/glass and Au) (Fig. S6†). By contrast, the coexistence of ohmic (CQD/Au) and Schottky (CQD/FTO-glass) contacts caused nonlinear I - V behavior in the CQD single-layer device (Fig. S7†). Consequently, the absence of a PEI structure in the device with Schottky barriers restricts the flow of electrons from the semiconductor to the electrode and lowers the photocurrent (Fig. S5c†).

The number of charge carriers in heterojunction devices is significantly influenced by the thickness of the photoactive layer, which in turn affects the performance of the device. Fig. 5a, b and S8† present the photocurrent and responsivity of the device prepared by spin-coating as a function of the concentration of PEI and CQD (PEI: 1, 3, 5, 7, and 9 mg mL⁻¹, and CQDs: 20, 30, 40, 50, and 60 wt%). Another essential metric for PDs is detectivity (D^*), which indicates the minimum detectable light intensity from the device. D^* is defined as follows:

$$D^* = \frac{R}{\sqrt{2e \times J_{\text{dark}}}} \quad (\text{Jones})$$

where e is the elemental charge, R is the responsivity, and J_{dark} is the dark-current density. When the PEI layer is thin (PEI concentrations: 1 and 3 mg mL⁻¹), the light of 254 nm wavelength cannot reach the PEI layer, partly because of the thick CQD layer, partly because the increased electron transport pathways in the absorption region leads to an unbalanced PEI/CQD interface (direction of incident light: top Au electrode → PEDOT:PSS → CQDs → bottom PEI). Moreover, CQDs have inherently high electronic conductivity, which contributes to the instability of their photoelectronics. When CQDs are exposed to radiation, the electrons trapped in the intrinsic vacancies and grain boundaries can be instantly released, particularly from the unstable PEI/CQD interface. The photocurrent increased excessively and momentarily because the electrons flew to the bottom electrode. Therefore, at concentrations of 1 and 3 mg mL⁻¹, the device did not receive a signal under UVC light. However, when the PEI concentration was gradually increased, the photoelectronic conductivity gradually became more stable owing to a decrease in the electronic conductivity (Fig. S9†) and the increase in the absorption of the PEI layer (maximum absorbance peak at 255 nm with a bandgap of 3.2 eV), as shown in Fig. S10.† The signal generated by the device gradually increased and reached the maximum at a PEI concentration of 7 mg mL⁻¹, with a photocurrent of 180 nA, responsivity of 3.6 mA W⁻¹, and detectivity of 1.4×10^{10} Jones (Fig. S8a†). By contrast, when the thickness of the PEI layer exceeded the threshold (9 mg mL⁻¹), the intensity of the 254 nm light reaching the absorption region was reduced owing to the excessively thick PEI layer, decreasing the efficiency of the PD device. The thickness of the CQD layer changed when the concentration was decreased from 60 to 20 wt% at a fixed PEI concentration (5 mg mL⁻¹). A further decrease in the film thickness was accompanied by an increase in the photocurrent,

leading to an increase in responsivity and detectivity (Fig. S8b†). In contrast, the PD performance deteriorated when the thickness of the CQD film decreased because the electrons with shortened traveling distances were less likely to be trapped in the O vacancies and grain boundaries in the thicker PEI film. The best PD performance was achieved with CQD-30, which afforded a photocurrent of 360 nA, responsivity of 0.78 mA W⁻¹, and detectivity of 5.1×10^{10} Jones. A further decrease in the concentration of the CQD solution (CQD-20) reduced the responsivity and photocurrent owing to the significant decrease in the number of photovoltaic cells. However, the worst performance of CQD-30 can be explained by the thick CQD film, which leads to many ineffective electrons trapped in the O vacancies and grain boundaries.

In short, the photocurrent depends on the concentrations of PEI and CQD solution. More importantly, the photocurrent development mechanism is governed by the properties and thicknesses of the films in the PD device. PEI acts as a buffer layer, stabilizing the CQD and FTO electrode interface. Increasing the PEI thickness effectively controls the electrons trapped in the intrinsic vacancies and grain boundaries that can be instantly released from the PEI/CQD interface when the CQDs are exposed to radiation. However, a PEI film with excessive thickness can reduce the 254 nm light intensity in the absorption region. In contrast, CQDs act as a potential photoactive layer. A thinner CQD film shortens the travel distance of electrons, and fewer electrons are trapped in O vacancies and grain boundaries. However, excessive photocell reduction due to the excessively thin CQD layer will reduce the photocurrent in

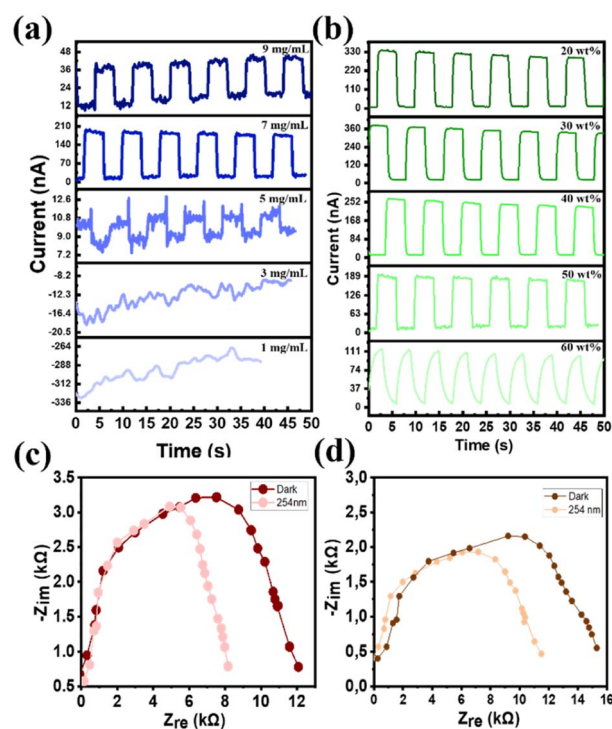


Fig. 5 Photocurrent of (a) PEI at different solution concentrations, (b) CQDs at different concentrations. Nyquist plots of (c) PEI (solution concentration: 7 mg mL⁻¹) and (d) CQDs (concentration: 30 wt%).



PEI films, deteriorating the PD performance. Therefore, adjusting the thickness of the PEI layer is necessary for optimizing the effective transport path in photovoltaic cells. Thus, the optimal conditions for the CQD and PEI layers were expected to produce a reasonably high photocurrent with a fast response. Under optimal conditions, the device can generate the highest on/off current ratio to achieve self-powered operation without consuming external energy.

Similar phenomena (brief spikes during pulsed light) have been documented for several optoelectronic devices, such as solar cells and PDs. Fig. 5c and d show the Nyquist plots of PEI and CQD films when the devices were illuminated by a light source with a wavelength of 254 nm at zero bias and in complete darkness. The PEI and CQD films were prepared from 7 mg mL⁻¹ PEI and 30 wt% CQD solutions, respectively (Plots for

each concentration are displayed individually in Fig. S11†). The radius of the semicircle in the Nyquist plot represents the charge-transfer resistance (R_{ct}). With light illumination, the number of photocarriers increased in the absorbance zone, reducing the radius of the semicircle. Because the trap sites in the thicker films enhanced the recombination of the electrons and holes, the radii of the semicircles in Fig. S10† generally increased as the concentration of the solution increased. Specifically, light with a wavelength of 254 nm could not reach the absorbance zone when the thickness of the PEI layer was excessively high (PEI concentration 9 mg mL⁻¹). Fewer electron-hole pairs are generated in the narrow absorbance zone of a thinner film (PEI-1 mg mL⁻¹ and CQD-60 wt%); therefore, the R_{ct} is low, and the reduction in R_{ct} caused by light exposure was not noteworthy. On the contrary, the reduction in R_{ct} due to

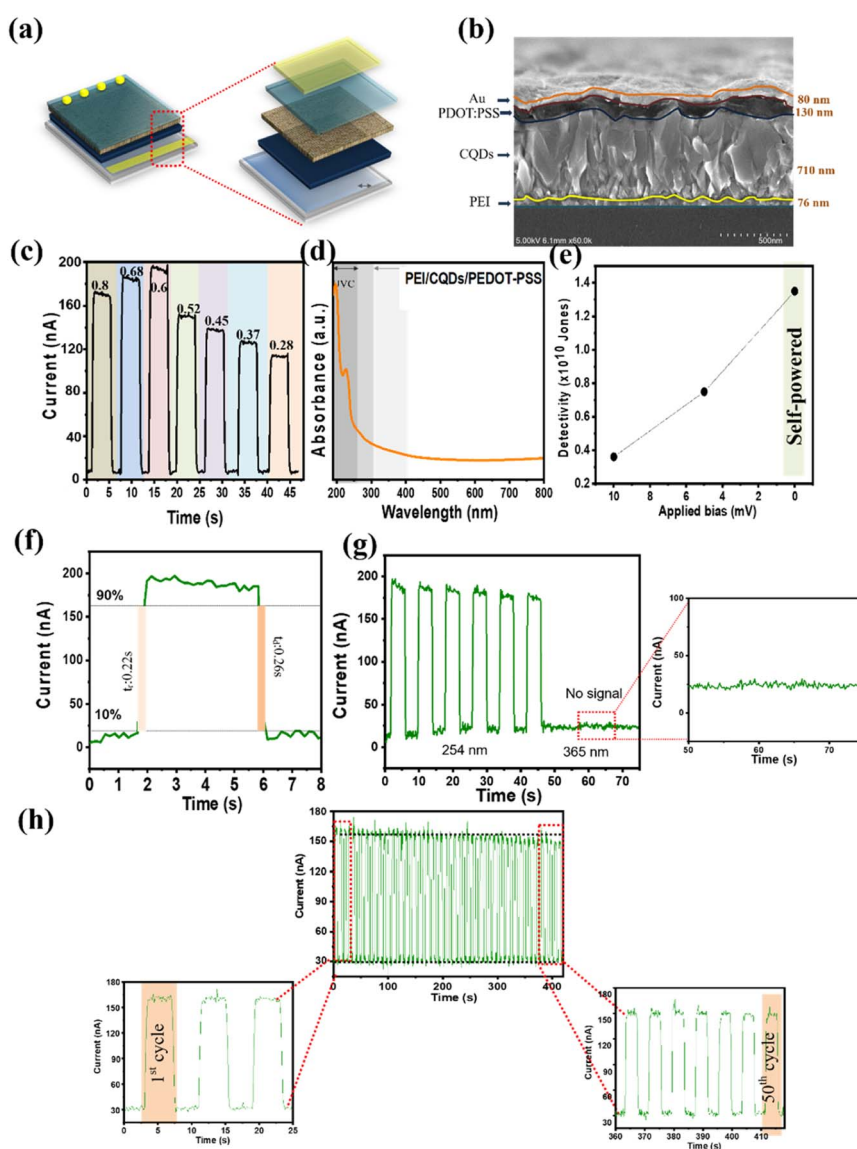


Fig. 6 (a) Schematic of UVC PD structure, (b) cross-sectional SEM image of the optimized PD, (c) photocurrent under different light intensities at zero bias, (d) spectral responsivity of the self-powered device, (e) detectivity of PD under different applied bias, (f) rise and decay times of the PD, (g) photocurrent versus time for PD under UVC and UVA illumination with an intensity of 0.6 mW cm⁻², and (h) long-term performance (50 cycles) when the PD was subjected to repeated on/off cycles (4 s of illumination per cycle).



light exposure was substantial for moderate film thicknesses (PEI-7 mg mL⁻¹ and CQD-30 wt%), suggesting that a significant number of photocarriers was produced in the absorbance area. Consequently, impedance spectroscopy (IS) analysis validated that the optimal solution concentrations for PEI and CQD were 7 mg mL⁻¹ and 30 wt%, respectively.

3.4. Performance of PEI-CQD heterojunction UVC PD

Fig. 6a schematically illustrates the vertical device structure, and Fig. 6b shows the cross-sectional SEM image of the optimized heterojunction PD (PEI-7/CQDs-30). The vertically stacked individual layers were clearly identified with well-defined thicknesses (Fig. 6b). In general, the boundaries of all organic materials are clearly shown. The thicknesses of the PEI, CQDs, PEDOT:PSS, and Au film layers were 76, 710, 130, and 80 nm, respectively. The *I*-*t* curves at different light intensities and bias voltages when exposed to 254 nm light are shown in Fig. 6c. Owing to the production of more photoelectrons at the PEI/CQD interface, the photocurrent increased from 160.8 to 190.8 nA at zero bias as the light intensity increased from 0.28 to 0.6 mW cm⁻². Thus, the best-performed CQD-based PD device reaches a photocurrent of 190 nA, responsivity of 1.5 mA W⁻¹, and detectivity of 1.8 × 10¹⁰ Jones under 0.6 mW cm⁻² (as shown in Fig. 6c and S12†). Regardless of the bias voltage, a steady drop in the responsivity was also observed as the light intensity increased from 0.6 to 0.8 mW cm⁻². This is explained by the non-linear increase in the on/off ratio with an increase in the light intensity. The optical-band selectivity of the PD was

determined by evaluating its spectral responsivity. Similar trends were observed under different bias settings (Fig. S13†). By contrast, a higher bias voltage resulted in a notably higher dark current and a lower on/off ratio. Notably, for bias voltages of 0, 5, and 10 mV, the dark currents were 19.86, 114.1, and 218.9 nA, respectively. The movement of charge carriers trapped in the defects of the CQD layer, and the increased drift speed caused a large dark current under bias conditions. The spectral responsiveness in the 220–800 nm wavelength range is displayed in Fig. 6d. The *D** exhibited a similar trend to that of responsivity, as evidenced by its negative correlation with both bias voltage and light intensity (Fig. 6e).

Because of the low dark current density, high detectivity of 1.3 × 10¹⁰ Jones was observed at zero bias. Consequently, the self-powered operation could improve detectivity and responsivity. The rise (time required for the maximum current to increase from 10% to 90%) or decay (time required for maximum current to decrease from 90% to 10%) period is referred to as the response time (*R*_r). The heterojunction PD exposed to 4 s light pulses had rise and decay times of 0.22 and 0.26 s, respectively, as shown in Fig. 6f. The CQD-based PD demonstrated high efficiency under irradiation at 254 nm compared with that under UV B-band (UVB) and UV A-band (UVA) light (no signal) (Fig. 6g and S14†). The applicability of PD sensors largely depends on the stability and continuity of the PDs. The photocurrent remained almost constant for 50 on/off light cycles, as illustrated in Fig. 6h (4 s of illumination for each pulse), indicating the consistent and reliable performance of the PD over an extended period of use.

Table 2 Comparison of the performance of PEI-CQD UVC PD with that of organic-based UV PDs reported in the literature^a

Photo-absorbing materials	Voltage (V)	Wavelength (nm)	Responsivity (mA W ⁻¹)	Detectivity (Jones)	Thickness (nm)	Ref.
PC ₆₁ BM	-1.0	500	0.03	—	—	86
PC ₆₁ BM	-1.0	525	0.06	1.00 × 10 ¹¹	120	87
Cy7-T:C ₆₀	-1.0	850	0.12	1.00 × 10 ¹²	30	88
1(Pyrl):C ₆₀	0	481	0.07	2.00 × 10 ¹¹	76	89
PolyTPD:SBTDIC	0	740	0.06	1.42 × 10 ¹³	70	40
PCZ-Th-DPP	-1.0	709	—	4.63 × 10 ¹²	300	90
Rubrene:C ₆₀	-1.0	470	0.21	—	70	91
M1:C ₆₀	-3.0	550	0.26	3.73 × 10 ¹³	85	92
CiInPc:C ₆₀	-1.0	705	0.45	3.30 × 10 ¹²	120	93
1a:C ₆₀	-3.0	560	0.32	4.37 × 10 ¹³	85	94
SubPc:C ₆₀	-1.0	600	0.08	7.50 × 10 ¹¹	—	95
DMQA:SubPc	-5.0	586	0.27	2.34 × 10 ¹²	90	96
DM-2,9-DMQA:SubPc	-3.0	580	0.56	2.03 × 10 ¹²	70	97
PCPDTBT:ZnO	—	725	0.40	—	70	98
PSBOTz:PNDBO	-2.0	530	0.07	1.10 × 10 ¹³	—	99
1(Pyrl):1(Hex):C ₆₀	0	754	0.086	1.10 × 10 ¹⁰	40	100
CQDs	0	254	1.5	1.8 × 10¹⁰	~700	This study

^a PC₆₁BM - [6,6]-phenyl-C₆₁-butyric acid methyl; Cy7-T:C₆₀ - heptamethine cyanine dyes/C₆₀ fullerene; 1(Pyrl):C₆₀ - merocyanine/C₆₀; polyTPD:SBTDIC - poly(*N,N*-bis-4-butylphenyl-*N,N'*-bisphenyl)benzidine:(4,4,10,10-tetrakis(4-hexylphenyl)-5,11-(*n*-hexylthio)-4,10-dihydrodithienyl[1,2-*b*:4,5-*b'*]benzodithiophene-2,8-diyl)bis(2-(3-oxo-2,3-dihydroinden-1-ylidene)malononitrile); PCZ-Th-DPP - phenanthrocarbazole:diketopyrrolopyrrole; M1:C₆₀ - M-molecules and C₆₀; CiInPc:C₆₀ - indium(iii) phthalocyanine chloride:C₆₀; 1a:C₆₀ - dipolar donor-acceptor (1a) molecules:C₆₀; SubPc:C₆₀ - boron subphthalocyanine chloride:C₆₀; DMQA:SubPc - *N,N*-dimethyl quinacridone:SubPc; DM-2,9-DMQA:SubPc - 2,9-dimethylquinacridone (2,9-DMQA):SubPc; PCPDTBT:ZnO - poly[2,6-(4,4-bis(2-ethylhexyl)-4*H*-cyclopenta [2,1-*b*; 3,4-*b'*]dithiophene)-*alt*-4,7(2,1,3-benzothiadiazole):ZnO]; PSBOTz:PNDBO - (4,8-bis(5-((2-ethylhexyl)thio)thiophen-2-yl)benzo[1,2-*b*:4,5-*b'*]dithiophene-2,6-diyl-*alt*-2,5-bis(5-bromo-4-octylthiazol-2-yl)thiophene-2,5-diyl)-*N,N'*-bis(2-ethylhexyl)-1,8,4,5-phthalenetetracarboxydiimide-2,6-diyl-*alt*-1,8-octyldioxy-4,4-diphenyl; 1(Pyrl):1(Hex):C₆₀ - 2-[4-(*tert*-butyl)thiazol-2(3*H*)-ylidene]malononitrile:2-aminothiophene:C₆₀.



Table 2 summarizes the performance of contemporary UV PDs. Compared with the performance of other recently reported UV PDs, the PD reported in this study exhibited high responsivity (1.5 mA W^{-1}) and detectivity (1.8×10^{10} Jones) at 254 nm during self-powered operation. In comparison to the performance of other recent photoactive materials, our PEI-CQD exhibits high responsibility, many times higher than reported organic-based UV devices, along with self-powering ability as another advantage in this research. In summary, this work proposes a combined strategy to optimize PEI thickness for stabilizing the PEI/CQD interface and to use organic materials that have inherently high electronic conductivity with appropriate film thickness for enhancing PD performance. The optimal PEI thickness achieves stable electron mobility across the PEI/CQD interface ($3.3 \times 10^{-6} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$). CQD films improve the photocurrent and responsibility to 360 nA and 1.5 mA W^{-1} , respectively. Therefore, this discovery demonstrated a simple and effective process to achieve high-performance electrodes for optoelectronics, especially deep UV photodetectors.

4. Conclusion

An all-organic-based, high-performance, self-powered PEI-CQD heterojunction UVC PD was developed through a simple, low-temperature solution process using natural precursors and a one-step ultrasonic reaction procedure. The CQDs were characterized by SEM, UV-vis absorbance spectroscopy, XRD, and FT-IR spectroscopy. The layer thicknesses of the PEI film and CQD film were optimized to achieve the best device performance. Under the optimal conditions, the PEI/CQD PD exhibits a strong photocurrent (1.5 mA W^{-1} at 254 nm), high detectability (1.8×10^{10} Jones at 254 nm), an ultrafast response speed, and long-term stability during the power supply sequence. Overall, the PEI/CQD PD reported in this study paves the way for the development of highly efficient, inexpensive, and form-factor UVC sensing devices for various applications.

Data availability

The authors do not have permission to share data.

Author contributions

Vo Pham Hoang Huy: data curation, writing – original draft, investigation. Chung Wung Bark: conceptualization, methodology, writing – review & editing, supervision, funding acquisition.

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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