


Cite this: *RSC Adv.*, 2024, 14, 4966

# Two-dimensional Janus Si<sub>2</sub>OX (X = S, Se, Te) monolayers as auxetic semiconductors: theoretical prediction

Nguyen P. Q. Anh,<sup>a</sup> N. A. Poklonski,<sup>b</sup> Vo T. T. Vi,<sup>c</sup> Cuong Q. Nguyen<sup>d,e</sup> and Nguyen N. Hieu<sup>d,\*</sup>

The auxetic materials have exotic mechanical properties compared to conventional materials, such as higher indentation resistance, more superior sound absorption performance. Although the auxetic behavior has also been observed in two-dimensional (2D) nanomaterials, to date there has not been much research on auxetic materials in the vertical asymmetric Janus 2D layered structures. In this paper, we explore the mechanical, electronic, and transport characteristics of Janus Si<sub>2</sub>OX (X = S, Se, Te) monolayers by first-principle calculations. Except for the Si<sub>2</sub>OTe monolayer, both Si<sub>2</sub>OS and Si<sub>2</sub>OSe are found to be stable. Most importantly, both Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are predicted to be auxetic semiconductors with a large negative Poisson's ratio. The auxetic behavior is clearly observed in the Janus Si<sub>2</sub>OS monolayer with an extremely large negative Poisson's ratio of −0.234 in the *x* axis. At the equilibrium state, both Si<sub>2</sub>OS and Si<sub>2</sub>OSe materials exhibit indirect semiconducting characteristics and their band gaps can be easily altered by the mechanical strain. More interestingly, the indirect–direct bandgap phase transitions are observed in both Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers when the biaxial strains are introduced. Further, the studied Janus structures also exhibit remarkably high electron mobility, particularly along the *x* direction. Our findings demonstrate that Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are new auxetic materials with asymmetric structures and show their great promise in electronic and nanomechanical applications.

Received 30th January 2024  
Accepted 31st January 2024

DOI: 10.1039/d4ra00767k

rsc.li/rsc-advances

## 1 Introduction

The auxetic material is a material possessing a negative Poisson's ratio (NPR). The material contracts laterally when compressed and expands laterally when stretched. NPR has been observed in both bulk and low-dimensional materials. For example, perovskite,<sup>1</sup> honeycombs,<sup>2</sup> ceramics,<sup>3</sup> and cubic metals<sup>4</sup> are bulk materials possessing the NPR phenomena. NPR has been also found in one-dimensional (1D) structures.<sup>5,6</sup> Particularly, both normal and auxetic behaviors coexist in 1D poly[5]asterane.<sup>7</sup> Recently, the auxetic effects have also been revealed in two-dimensional (2D) structures. Black phosphorus is the first 2D material reported to have an NPR with a small value of −0.027.<sup>8</sup> Later, the auxetic behavior has also been found in other 2D layered nanomaterials such as borophene,<sup>9</sup>

Be<sub>5</sub>C<sub>2</sub>,<sup>10</sup> h-BN,<sup>11</sup> and 1T phase of transition metal dichalcogenides.<sup>12</sup> In parallel with the bulk and 2D materials, the auxetic effect is also explored in van der Waals superlattices.<sup>13</sup> The graphene/hexagonal-BN superlattice has an NPR with values of −0.109 and −0.111 depending on its atomic stacking configuration.<sup>13</sup> The origin of auxetic phenomena is found to be the structure of the material. It can be explained by the wrinkled or re-entrant crystal structure. The auxetic materials have exotic mechanical properties compared to conventional materials, such as higher indentation resistance,<sup>14</sup> more superior sound absorption performance.<sup>15</sup> Therefore, NPR materials have application prospects in the fields of industry,<sup>16</sup> biomedicine,<sup>17</sup> and sensors.<sup>18</sup>

Up to now, the variety and application prospects of 2D materials have been extensively demonstrated.<sup>19–21</sup> Among them, the SiS monolayer is a typical compound for 2D structures with extraordinary physical characteristics. Particularly, 2D SiS layers can be stable in different polymorphs, including *Pma2*-SiS, *Pmma*-SiS, and silicene sulfide.<sup>22</sup> The SiS monolayer was found to exhibit mechanical, energetic, and thermal stability.<sup>23</sup> Besides, the SiS monolayer possesses excellent mechanical and electronic properties. The SiS exhibits semiconducting features with a tunable direct energy gap and directionally anisotropic carrier mobility. The mobility of

<sup>a</sup>Faculty of Electrical, Electronics and Materials Technology, University of Sciences, Hue University, Hue, 530000, Viet Nam

<sup>b</sup>Faculty of Physics, Belarusian State University, Minsk 220006, Belarus

<sup>c</sup>Faculty of Basic Sciences, University of Medicine and Pharmacy, Hue University, Hue 530000, Viet Nam

<sup>d</sup>Institute of Research and Development, Duy Tan University, Da Nang 550000, Viet Nam. E-mail: hieunm@duytan.edu.vn

<sup>e</sup>Faculty of Natural Sciences, Duy Tan University, Da Nang 550000, Viet Nam



electrons in the SiS structure is superior with a value of  $1.11 \times 10^4 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . Most importantly, SiS exhibits high NPR with a value of  $-0.19$  in the  $x$  axis.<sup>23</sup> Therefore, the SiS monolayer has great promise in applications for nanomechanics and optoelectronics.<sup>23</sup>

Since the experimental synthesis of 2D vertically asymmetric MoSSe sheet,<sup>24,25</sup> 2D Janus structures have become an important object in the studies in the material science field.<sup>26–29</sup> The 2D Janus monolayers exhibit many exotic physical properties compared to their original symmetric counterparts.<sup>30–34</sup> Recently, a few works have presented a particularly intriguing property of the Janus structures, namely the auxetic behavior. The Poisson's ratio with a negative value has been found in both in-plane directions. For example, the  $\text{Si}_2\text{SSe}$  monolayer has NPR with a value of  $-0.11$  ( $-0.08$ ) along the  $x$  ( $y$ ) axis.<sup>23</sup> In term of  $\text{Si}_2\text{XY}$  ( $X/Y = \text{S, Se, Te}$ ) structures, NPR values lie between  $-0.062$  and  $-0.113$  ( $-0.082$  and  $-0.122$ ) along the  $x$  ( $y$ ) axis.<sup>35</sup> However, little research on the NPR effect has been performed on asymmetric Janus structures. Therefore, the search for Janus structures with auxetic behavior is very important. Motivated by these problems, we report the mechanical features, electronic characteristics, and carrier mobility of Janus 2D  $\text{Si}_2\text{OX}$  ( $X = \text{S, Se, Te}$ ) monolayers using density functional theory (DFT) method. It is demonstrated that the  $\text{Si}_2\text{OX}$  monolayers, except for  $\text{Si}_2\text{OTe}$ , are semiconductors with stable structures, high carrier mobility, and strain-controlled band gap. Most importantly, the NPR phenomena are explored in the Janus  $\text{Si}_2\text{OX}$  monolayers. With these findings, we hope to provide fundamental insights and promote extensive studies for Janus structures as well as auxetic materials.

## 2 Theoretical model and computational details

Our first-principles simulations were carried out based on DFT technique by using the Vienna *ab initio* simulation package.<sup>36,37</sup> We used the projector augmented wave (PAW) method to investigate the core-electron interactions.<sup>36</sup> Generalized gradient approximation (GGA) parameterized by Perdew, Burke, and Ernzerhof (PBE)<sup>38</sup> was performed for the crystal structure optimizations. Dipole correction is included to treat the errors induced by the periodic boundary condition.<sup>39</sup> The Brillouin zone of the studied structures was sampled with a  $15 \times 15 \times 1$   $k$ -mesh by Monkhorst-Pack technique.<sup>40</sup> The GGA-PBE functional was used for the optimization of the crystal structures. The crystal structures were totally relaxed until the Hellman-Feynman force was less than  $1 \times 10^{-3} \text{ eV } \text{\AA}^{-1}$  and the total energy was converged to  $10^{-8} \text{ eV}$ . An energy cut-off of 500 eV was selected for the plane-wave basis functions. The GGA-PBE method was used to describe the electron exchange-correlation interaction. Also, the hybrid functional by Heyd-Scuseria-Ernzerhof (HSE06) was used to correct the band gap errors of the considered structures.<sup>41</sup> The van der Waals corrections were treated by using the DFT-D3 method suggested by Grimme.<sup>42</sup> To avoid the periodic interactions in the vertical direction, we inserted a vacuum distance of 20 Å along the  $z$  axis. The phonon

spectra of the studied materials were calculated by using the PHONOPY code<sup>43</sup> based on the finite displacement technique. The *ab initio* molecular dynamics (AIMD) simulations with the NVT ensemble (the number of particles  $N$ , the calculated cell volume  $V$ , and temperature  $T$  were fixed) were used to investigate the thermodynamic stabilities of the considered materials.<sup>44</sup> The deformation potential (DP) method was adopted to compute the carrier mobilities.<sup>45</sup>

## 3 Results and discussion

### 3.1 Crystal structure, stability, and mechanical characteristics

The crystal structures of the vertical asymmetric Janus  $\text{Si}_2\text{OX}$  ( $X = \text{S, Se, and Te}$ ) monolayer are shown in Fig. 1. The Janus  $\text{Si}_2\text{OX}$  monolayers belong to  $Pma2$  symmetry. We can construct the Janus  $\text{Si}_2\text{OX}$  from the  $Pma2$ - $\text{SiX}$  monolayer<sup>22</sup> by substituting the bottom chalcogen layer  $X$  with the oxygen layer as presented in Fig. 1. We can see that the unit cell of the  $\text{Si}_2\text{OX}$  monolayers contains 8 atoms, including four Si, two O, and two  $X$  atoms, as shown in Fig. 1(a). It is found that Janus  $\text{Si}_2\text{OX}$  monolayers possess an anisotropic crystal structure. The lattice constants  $a(b)$  of  $\text{Si}_2\text{OS}$ ,  $\text{Si}_2\text{OSe}$ , and  $\text{Si}_2\text{OTe}$  are calculated to be 5.92(3.90), 5.92(3.92), and 5.97(4.03) Å, respectively. It is shown that there is no significant different in bond angle  $\varphi_{\angle \text{Si-O-Si}}$  between the structures. Meanwhile, the large different in  $\varphi_{\angle \text{Si-X-Si}}$  between the structures is found. The obtained  $\varphi_{\angle \text{Si-X-Si}}$  is found to be  $87.82^\circ$ ,  $80.37^\circ$ , and  $72.23^\circ$  for  $\text{Si}_2\text{OS}$ ,  $\text{Si}_2\text{OSe}$ , and  $\text{Si}_2\text{OTe}$ , respectively. We present the structural parameters of  $\text{Si}_2\text{OX}$  in Table 1.

To demonstrate the stability under normal conditions of  $\text{Si}_2\text{OX}$  materials, we examine their phonon dispersions. Fig. 2 presents the phonon spectra of three  $\text{Si}_2\text{OX}$  monolayers in the first Brillouin zone. Since the unitcell of  $\text{Si}_2\text{OX}$  contains 8 atoms, there are 24 vibrational branches within their vibrational spectra. It can be found that there are three acoustic and 21

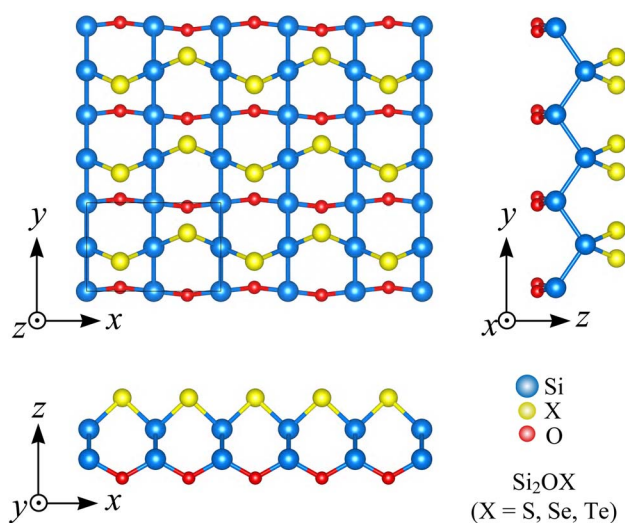
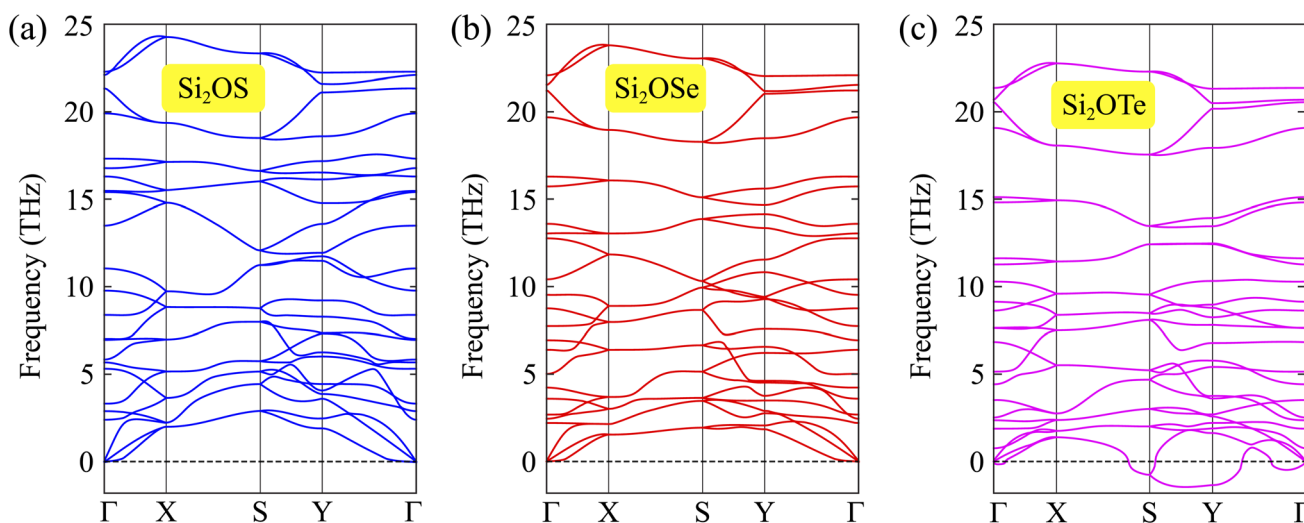


Fig. 1 Optimized crystal structures of  $\text{Si}_2\text{OX}$  monolayer in different views. The unitcell is denoted by the rectangle in the top view image.



**Table 1** Optimized lattice constants  $a$  and  $b$ , chemical bond lengths  $d$ , bond angle  $\phi$ , and elastic constants  $C_{ij}$  of Janus  $\text{Si}_2\text{OX}$  systems. Since the Janus  $\text{Si}_2\text{OTe}$  is unstable, the calculations for  $C_{ij}$  of  $\text{Si}_2\text{OTe}$  is excluded

	$a$ (Å)	$b$ (Å)	$d_{\text{Si-O}}$ (Å)	$d_{\text{Si-X}}$ (Å)	$d_{\text{Si-Si}}$ (Å)	$\phi_{\angle \text{Si-O-Si}}$ (°)	$\phi_{\angle \text{Si-X-Si}}$ (°)	$C_{11}$ (N m <sup>-1</sup> )	$C_{12}$ (N m <sup>-1</sup> )	$C_{22}$ (N m <sup>-1</sup> )	$C_{66}$ (N m <sup>-1</sup> )
$\text{Si}_2\text{OS}$	5.92	3.90	1.71	2.13	2.36	120.29	87.82	92.34	-14.56	62.75	14.70
$\text{Si}_2\text{OSe}$	5.92	3.92	1.71	2.29	2.36	120.22	80.37	94.48	-5.79	69.99	18.00
$\text{Si}_2\text{SeTe}$	5.97	4.03	1.71	2.53	2.38	120.35	72.23	—	—	—	—



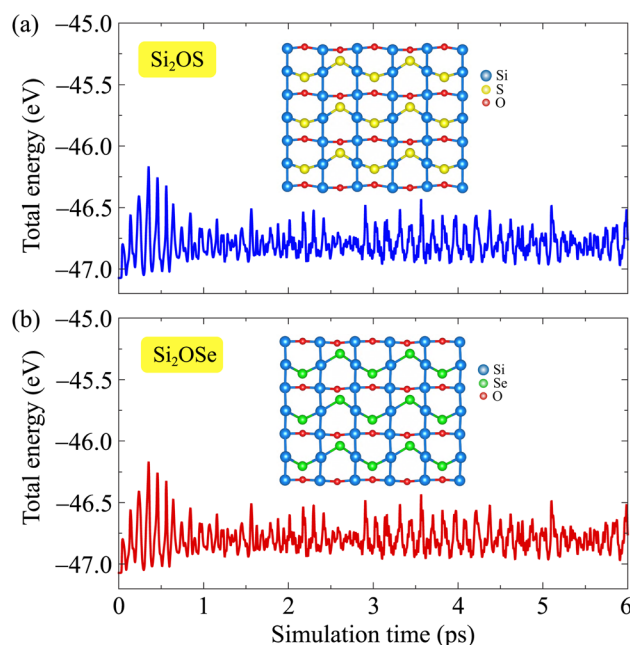
**Fig. 2** Phonon dispersions along the  $\Gamma$ -X-S-Y- $\Gamma$  high-symmetry line of  $\text{Si}_2\text{OS}$  (a),  $\text{Si}_2\text{OSe}$  (b), and  $\text{Si}_2\text{OTe}$  (c).

optical vibrational branches in the phonon dispersion. Besides, the frequencies of the vibrational branches of the two Janus  $\text{Si}_2\text{OS}$  and  $\text{Si}_2\text{OSe}$  are positive, while the negative frequencies are found in the vibrational spectrum of  $\text{Si}_2\text{OTe}$  as revealed in Fig. 2(c). Once the phonon dispersions contain the negative frequencies, the restoring forces against the displacement of the atoms are suppressed. As a consequence, the crystal structure of these materials is unstable. It implies that two monolayers  $\text{Si}_2\text{OS}$  and  $\text{Si}_2\text{OSe}$  are dynamically stable, while the Janus  $\text{Si}_2\text{OTe}$  monolayer is unstable. Therefore, from here on, we only examine  $\text{Si}_2\text{OS}$  and  $\text{Si}_2\text{OSe}$  and exclude  $\text{Si}_2\text{OTe}$  from the calculations due to its structural instability.

Next, the thermal stability of these two monolayers is also investigated through the AIMD simulation. We perform the AIMD test within 6 ps (each step of 1 fs) at room temperature (300 K). The total energy fluctuations of  $\text{Si}_2\text{OS}$  and  $\text{Si}_2\text{OSe}$  as functions of simulation time are illustrated in Fig. 3. It is obvious that the total energies fluctuate only by a very small value. The atomic structures of the two monolayers are only slightly deformed and remain structurally solid. Neither the structural phase transitions nor the breaking of chemical bonds were found after the AIMD simulation. It suggests that both  $\text{Si}_2\text{OS}$  and  $\text{Si}_2\text{OSe}$  are thermodynamic stable at ambient temperature.

Further, we also evaluate the mechanical stabilities of Janus  $\text{Si}_2\text{OS}$  and  $\text{Si}_2\text{OSe}$  monolayers based on the analysis their elastic characteristics. We calculate the elastic constants of 2D materials, namely  $C_{11}$ ,  $C_{12}$ ,  $C_{22}$ , and  $C_{66}$ , by using Duerloo's

method.<sup>46</sup> Two in-plane directions of the Janus systems are subjected to small uniaxial strains from -1.5% to 1.5%. At the specified strain, the positions of atoms are re-optimized and the



**Fig. 3** AIMD calculations of  $\text{Si}_2\text{OS}$  (a) and  $\text{Si}_2\text{OSe}$  (b) systems. Inset presents the crystal structure at the end of the simulation progress.



strain-dependence of the energy is found. The calculated energies are polynomial fitted, which yields the in-plane stiffness coefficients. The in-plane stiffness coefficients of 2D Janus Si<sub>2</sub>OS are found to be  $C_{11} = 92.34 \text{ N m}^{-1}$  and  $C_{22} = 62.75 \text{ N m}^{-1}$ , which are close to those of Si<sub>2</sub>OSe being  $C_{11} = 94.48 \text{ N m}^{-1}$  and  $C_{22} = 69.99 \text{ N m}^{-1}$ , respectively. Particularly, both Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers possess a negative value for elastic constant  $C_{12}$  ( $C_{12} = -14.56 \text{ N m}^{-1}$  for Si<sub>2</sub>OS and  $C_{12} = -5.79 \text{ N m}^{-1}$  for Si<sub>2</sub>OSe), suggesting that unusual NPR may occur in these investigated monolayers.<sup>47,48</sup> From the obtained elastic constants as listed in Table 1, it is revealed that  $C_{ij}$  of both Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers satisfy the criteria proposed by Born for mechanical stabilities for 2D materials.<sup>48,49</sup> This indicates that the mechanical stability of Si<sub>2</sub>OS and Si<sub>2</sub>OSe is confirmed.

Next, we investigate the mechanical characteristics of Si<sub>2</sub>OS and Si<sub>2</sub>OSe through the calculations for Young's modulus and Poisson's ratio. The angular-dependent Young's modulus  $Y_{2D}(\theta)$  and Poisson's ratio  $\mathcal{P}(\theta)$  of the studied materials can be written by the following expressions:<sup>50,51</sup>

$$Y_{2D}(\theta) = \frac{C_{11}C_{22} - C_{12}^2}{C_{11}\sin^4\theta - \Omega\sin^2\theta\cos^2\theta + C_{22}\cos^4\theta}, \quad (1)$$

$$\mathcal{P}(\theta) = \frac{C_{12}(\sin^4\theta + \cos^4\theta) - \Pi\sin^2\theta\cos^2\theta}{C_{11}\sin^4\theta - \Omega\sin^2\theta\cos^2\theta + C_{22}\cos^4\theta}, \quad (2)$$

where  $\Omega = 2C_{12} - (C_{11}C_{22} - C_{12}^2)/C_{66}$ ,  $\Pi = C_{11} + C_{22} - (C_{11}C_{22} - C_{12}^2)/C_{66}$ , and  $\theta$  is the angle between the studied direction and the x-axis.

The angular-dependence of Young's modulus of the studied structures is depicted as shown in Fig. 4(a). Our calculated results revealed that  $Y_{2D}$  of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are greatly directionally anisotropic. This is attributed to the in-plane asymmetric geometric structure of the Janus Si<sub>2</sub>OX systems. The Si<sub>2</sub>OSe monolayer has a larger Young's modulus value than that of Si<sub>2</sub>OS monolayer. Young's modulus has the maximum value of  $94.00 \text{ N m}^{-1}$  ( $88.94 \text{ N m}^{-1}$ ) at  $\theta = 0^\circ$ , and the minimum value of  $48.35 \text{ N m}^{-1}$  ( $39.28 \text{ N m}^{-1}$ ) at  $\theta = 48^\circ$  for the Janus Si<sub>2</sub>OSe (Si<sub>2</sub>OS) compound, respectively. Possessing a small  $Y_{2D}$ , Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers have high mechanical flexibility compared to other 2D structures, such as graphene ( $344 \text{ N m}^{-1}$ ),<sup>52</sup> binary MoS<sub>2</sub> compound ( $130 \text{ N m}^{-1}$ )<sup>53</sup> or Janus ternary MoSSe monolayer ( $113 \text{ N m}^{-1}$ ).<sup>54</sup>

In Fig. 4(b), we reveal the calculated polar diagrams of Poisson's ratio  $\mathcal{P}(\theta)$  of Janus Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers. It is found that Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers exhibit a high anisotropy. Remarkably, the negative Poisson's ratio is found in both Si<sub>2</sub>OS and Si<sub>2</sub>OSe compounds. This is consistent with the expectation above that these structures have a negative  $C_{12}$ . The NPR feature can be seen in both  $x$  and  $y$  axes as depicted in Fig. 4(b). The calculated results demonstrate that Janus Si<sub>2</sub>OS has a large in-plane NPR, up to  $-0.234$  and  $-0.158$  in the  $x$  and  $y$  axes, respectively. Surprisingly, these NPR values are much larger than those of SiS ( $-0.19$  and  $-0.10$ )<sup>48</sup> and Janus Si<sub>2</sub>SSe ( $-0.131$  and  $-0.122$  in the  $x$  and  $y$  axes, respectively).<sup>35</sup> The NPR values of Si<sub>2</sub>OSe monolayer are found to be  $-0.083$  ( $-0.061$ ) in the  $x$ ( $y$ ) axis as presented in Fig. 4(b). The appearance of the NPR confirms that Janus Si<sub>2</sub>OS and Si<sub>2</sub>OSe can be treated as auxetic semiconductors. Compared with usual nanomaterials, 2D auxetic structures possess many extraordinary physical features with promising application in various fields of nanotechnology.<sup>55</sup>

### 3.2 Electronic properties

Here, we report the electronic properties of two stable structures of Si<sub>2</sub>OX, namely Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers. First, the band structures of the two compounds are examined by both the PBE and HSE06 methods as shown in Fig. 5. It can be observed that Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers exhibit indirect semiconducting characteristics. In the case of Si<sub>2</sub>OS, the conduction band minimum (CBM) and valence band maximum (VBM) are found at the  $\Gamma$  point and on the  $X\Gamma$  line, respectively. Meanwhile, the CBM and VBM of Si<sub>2</sub>OSe are found respectively at the Y and  $\Gamma$  points. It should be noted that the vertical symmetric SiS and SiSe are direct bandgap semiconductors.<sup>22,23</sup> Thus, the symmetry breaking significantly changes the electronic characteristics of Janus structures. At the PBE level, the indirect bandgaps of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are respectively obtained to be 1.35 and 1.20 eV. The bandgap correction by using the HSE06 functional shows that the bandgaps of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are 2.18 and 1.97 eV, respectively. The obtained HSE06 bandgaps of Si<sub>2</sub>OS and Si<sub>2</sub>OSe are wider than those of SiS (1.22 eV)<sup>23,56</sup> and SiSe (1.44 eV<sup>23</sup> and 1.35 eV<sup>56</sup>). The obtained bandgaps of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are listed in Table 2.

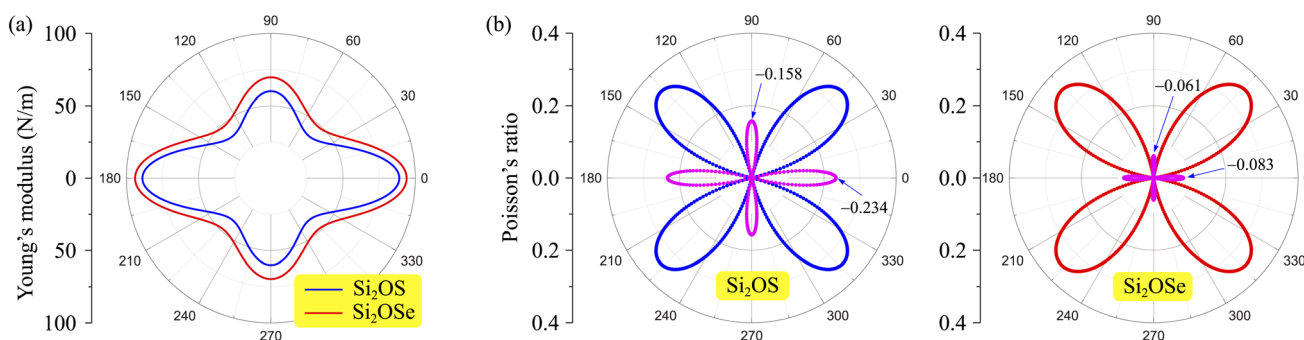


Fig. 4 (a) Young's modulus  $Y_{2D}(\theta)$  and (b) Poisson's ratio  $\mathcal{P}(\theta)$  of Janus Si<sub>2</sub>OS and Si<sub>2</sub>OSe structures. Pink symbols in (b) indicate the negative values of  $\mathcal{P}$ .



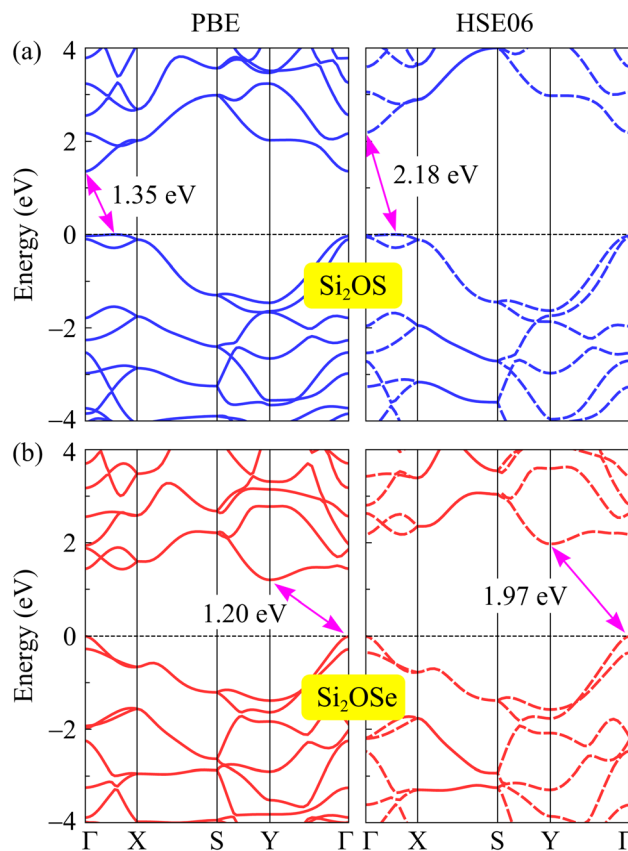


Fig. 5 The PBE and HSE06 band structures of Si<sub>2</sub>OS (a) and Si<sub>2</sub>OSe (b) monolayers.

In addition to the electronic band diagrams, the work function is also a fundamental electronic feature of the material that should be taken into account. The work function denotes the required energy for electrons to escape from the surfaces of the studied structures and can be estimated based on the vacuum level  $E_{\text{vac}}$  and Fermi level  $E_{\text{F}}$  as:  $\Phi = E_{\text{vac}} - E_{\text{F}}$ . We can find the vacuum level based on calculations for the electrostatic potential of the studied structures. An intrinsic electric field exists in the vertical direction of the Janus material due to the electronegativity difference between the two sides of this layered structure. This difference in electronegativity leads to the vacuum level difference between the two sides of materials. As a consequence, there is a work function difference on the two sides of the studied asymmetric structures. Since the electronegativity of O atom is greater than that of the chalcogen atom X, the intrinsic electric field in the Si<sub>2</sub>OX structures is directed from X to O. In Fig. 6, we show the electrostatic potential

Table 2 Obtained bandgaps  $E_{\text{g}}$  using PBE/HSE06 functional, difference between the vacuum levels  $\Delta\Phi$ , and work function on the O(X) side  $\Phi_{\text{O}}(\Phi_{\text{X}})$  of Si<sub>2</sub>OX materials

	$E_{\text{g}}^{\text{PBE}}$ (eV)	$E_{\text{g}}^{\text{HSE06}}$ (eV)	$\Delta\Phi$ (eV)	$\Phi_{\text{O}}$ (eV)	$\Phi_{\text{X}}$ (eV)
Si <sub>2</sub> OS	1.35	2.18	0.06	5.46	5.40
Si <sub>2</sub> OSe	1.20	1.97	0.47	5.22	4.75

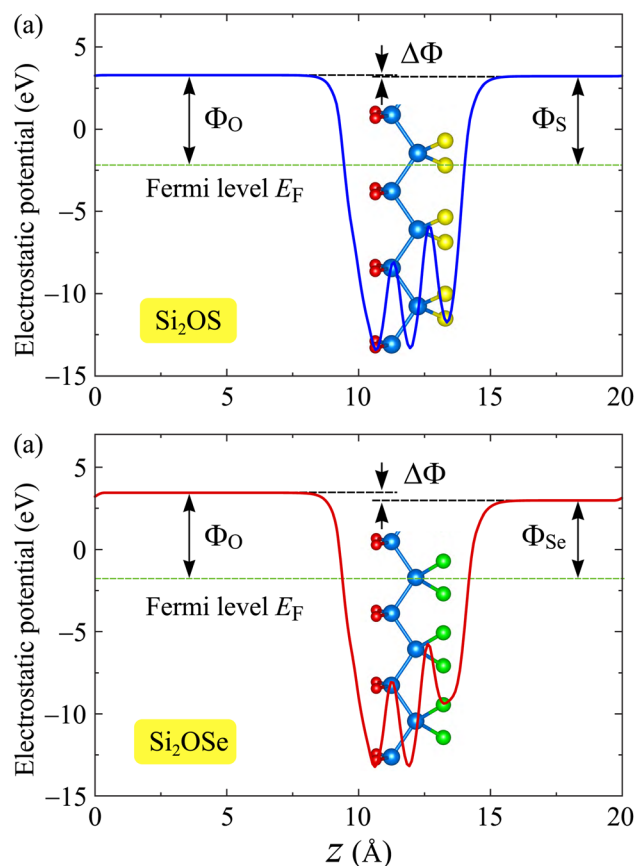


Fig. 6 Electrostatic potentials of Si<sub>2</sub>OS (a) and Si<sub>2</sub>OSe (b) monolayers.

energies with a dipole correction of Si<sub>2</sub>OS and Si<sub>2</sub>OSe materials. A vacuum level difference  $\Delta\Phi$  is found in both Si<sub>2</sub>OS and Si<sub>2</sub>OSe. The larger the difference in atomic size between O and X elements, the larger the value  $\Delta\Phi$  is. The calculated  $\Delta\Phi$  for Si<sub>2</sub>OS and Si<sub>2</sub>OSe are 0.06 and 0.47 eV, respectively. Table 2 summarizes our obtained results for  $\Delta\Phi$  and work functions of Si<sub>2</sub>OS and Si<sub>2</sub>OSe materials. The work functions of Si<sub>2</sub>OS are slightly higher than those of Si<sub>2</sub>OSe. The data in Table 2 reveal that it is easier for electrons to totally escape from the X surface than from the O surface.

It is well-known that there are many techniques to tune the electronic features of 2D materials, such as chemical functionalization, defect effects, doping, *etc.* Among these, mechanical strain is an effective and simple approach to control electronic properties.<sup>27,31,48</sup> In this paper, a biaxial strain between  $-6\%$  and  $6\%$  is applied to examine the strain-dependence of the electronic features of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers. The biaxial strain can be evaluated as  $\epsilon_{\text{b}} = (a - a_0)/a_0$ , where  $a_0$  and  $a$  denote the lattice constants of Si<sub>2</sub>OS and Si<sub>2</sub>OSe at equilibrium and strained states. Mechanical strains with positive and negative values correspond to tension and compression, respectively. Fig. 7 depicts the band structures of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers at several values of  $\epsilon_{\text{b}}$ . We can see that the mechanical strain significantly adjusts the electronic band structures, especially for the Janus Si<sub>2</sub>OS monolayer. When the tensile strain reaches  $+4\%$ , the indirect-direct



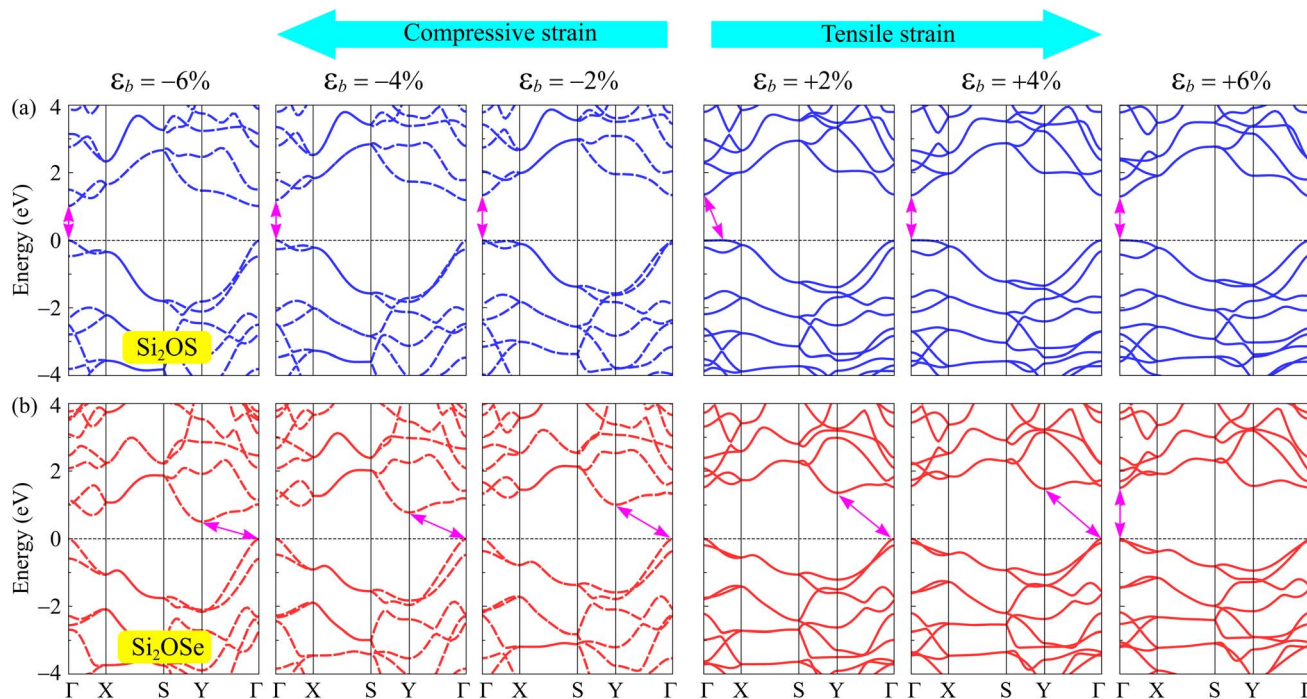


Fig. 7 The PBE band structures of (a) Si<sub>2</sub>OS and (b) Si<sub>2</sub>OSe materials under the biaxial strain  $\epsilon_b$ .

bandgap transition is found in the Si<sub>2</sub>OS structure as shown in Fig. 7(a). Also, Si<sub>2</sub>OS is found as direct semiconductor with both VBM and CBM at the  $\Gamma$  point when the tensile strain is introduced. Both tensile and compressive strains shift the band edges in the electronic structure of the Janus structure, leading to a change in the bandgap. Besides, the phase transition from indirect to direct bandgap is also observed in Si<sub>2</sub>OSe monolayer at  $\epsilon_b = 6\%$  and the applied strain significantly modulates its bandgap. Fig. 8 presents the bandgaps of Janus Si<sub>2</sub>OX structures as functions of  $\epsilon_b$ . As can be seen from Fig. 8, the bandgap of Si<sub>2</sub>OSe is significantly reduced by compressive strains, whereas tensile strain slightly increases its bandgap. In the case of Si<sub>2</sub>OS,

the biaxial strains reduce its bandgap. However, the effect of  $\epsilon_b$  on the energy gap of Si<sub>2</sub>OS is quite weak. The maximum bandgap of Si<sub>2</sub>OS is 1.35 eV at equilibrium ( $\epsilon_b = 0$ ) and the lowest is at  $\epsilon_b = -6\%$  with the value of 1.01 eV. The strain-induced phase transitions from indirect-to-direct energy gap in both Si<sub>2</sub>OS and Si<sub>2</sub>OSe make them promising for nano-electromechanical applications.

### 3.3 Carrier mobility

In the following, we compute the carrier mobilities in Janus Si<sub>2</sub>OS and Si<sub>2</sub>OXSe materials. Carrier mobility is a key parameter relevant to technological applications such as electronic devices. By employing the deformation potential (DP) approximation,<sup>45</sup> the mobilities of electrons and holes in 2D systems can be obtained by the following equation:<sup>57</sup>

$$\mu_{2D} = \frac{e\hbar^3 C_{2D}}{k_B T m^* \bar{m} E_d^2}, \quad (3)$$

where  $e$ ,  $\hbar$ ,  $k_B$ , and  $T$  refer to the elementary charge, the reduced Planck's constant, Boltzmann's constant, and the temperature, respectively.  $m^*$  and  $\bar{m} = \sqrt{m_x^* m_y^*}$  are respectively the carrier effective mass along the  $x/y$  axis and average effective mass.  $C_{2D}$  refers to the elastic modulus and  $E_d$  indicates the DP constant. Herein, the carrier mobility is calculated at  $T = 300$  K (room temperature). The effective mass of carriers, elastic modulus, and the DP constant are determined as follows:

$$\frac{1}{m^*} = \frac{1}{\hbar^2} \left| \frac{\partial^2 E(k)}{\partial k^2} \right|, \quad (4)$$

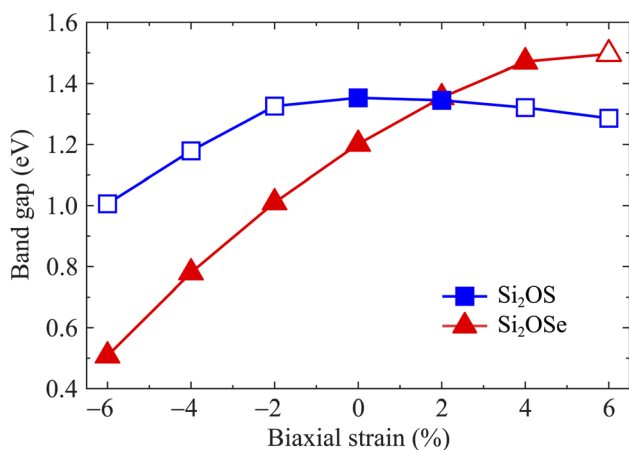


Fig. 8 Dependence of the bandgaps of Si<sub>2</sub>OS and Si<sub>2</sub>OSe on the biaxial strain  $\epsilon_b$ . The filled/empty shapes indicate the indirect/direct bandgaps, respectively.



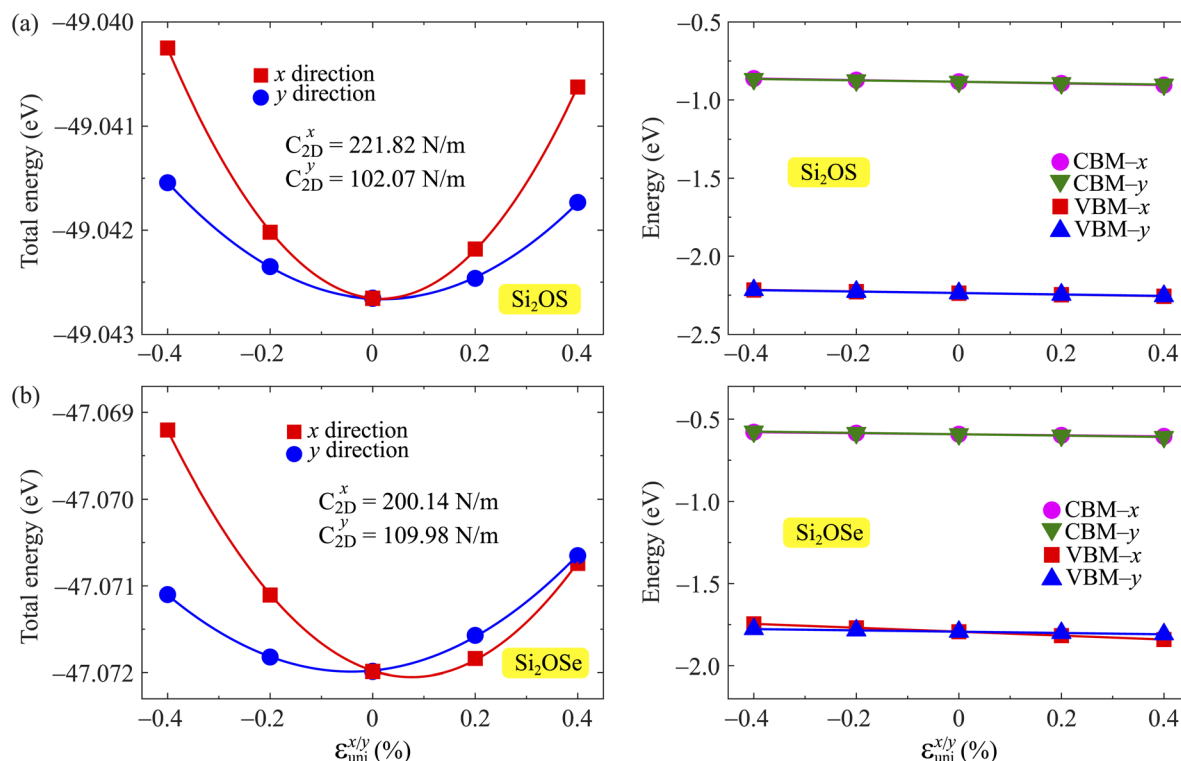


Fig. 9 The total energy and band-edge energy of Si<sub>2</sub>OS (a) and Si<sub>2</sub>OSe (b) monolayers as functions of small uniaxial strains  $\epsilon_{\text{uni}}^{x/y}$ .

$$C_{2D} = \frac{1}{V_0} \frac{\partial^2 E_{\text{tot}}}{\partial \epsilon_{\text{uni}}^2}, \quad (5)$$

$$E_d = \frac{\Delta E_{\text{edge}}}{\epsilon_{\text{uni}}}, \quad (6)$$

where  $E(k)$  indicates the wavenumber  $k$ -dependent energy at the VBM for holes and CBM for electrons.  $V_0$  and  $E_{\text{tot}}$  refer to the area of optimized unitcell and the total energy, respectively.  $\epsilon_{\text{uni}}$  is the small uniaxial strain along the  $x/y$  axis, which varies between 0 and  $\pm 0.4\%$  in our calculations for the transport parameters.  $\Delta E_{\text{edge}}$  stands for the uniaxial strain-induced energy change of the band edges. The transport parameters  $C_{2D}$  and  $E_d$  are obtained by fitting the strain-dependence of the total energies and band edges energy as presented in Fig. 9. The total energy depends not only strongly on the strain but also on the direction of the uniaxial strain, namely along the  $x$  or  $y$  axis. This will cause the elastic modulus of the examined structures to be high and they are high directional anisotropy. Meanwhile, the change in position of the band edges of Si<sub>2</sub>OS and Si<sub>2</sub>OSe

monolayers in the presence of the applied uniaxial strains along the  $x$  and  $y$  axes is small, suggesting that the directional isotropy of the DP constant is high.

The obtained transport parameters  $m^*$ ,  $C_{2D}$ ,  $E_d$ , and corresponding carrier mobility  $\mu_{2D}$  along the axes  $x$  and  $y$  are presented in Table 3. We can see that Janus Si<sub>2</sub>OS and Si<sub>2</sub>OSe exhibit highly directionally anisotropic transport features due to their in-plane anisotropic crystal lattice. The results of our calculations indicate that the electrons in Janus Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers have low effective mass. The electron effective masses of Si<sub>2</sub>OS and Si<sub>2</sub>OSe along the  $x(y)$  directions are respectively  $0.48m_0$  ( $0.50m_0$ ) and  $0.59m_0$  ( $1.39m_0$ ) with  $m_0$  being the free electron mass. Since the electrons have a low effective mass, they will respond quickly to the external field, which is expected to result in high electron mobilities in the Janus Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers. In Janus Si<sub>2</sub>OS, the electrons have a smaller effective mass than that of the holes. Meanwhile, the hole effective masses of Si<sub>2</sub>OSe are smaller than those of the electrons. These results are consistent with the band dispersion

Table 3 The calculated effective masses of carriers  $m^*$  (in unit of the mass of a free electron  $m_0$ ), elastic modulus  $C_{2D}$  (N m<sup>-1</sup>), DP constant  $E_d$  (eV), and carrier mobility  $\mu_{2D}$  (cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>) along the  $x/y$  transport axis of Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers

		$m_x^*$	$m_y^*$	$C_{2D}^x$	$C_{2D}^y$	$E_d^x$	$E_d^y$	$\mu_{2D}^x$	$\mu_{2D}^y$
Si <sub>2</sub> OS	Electron	0.48	0.50	221.82	102.07	-5.25	-4.47	727.98	443.60
	Hole	1.57	6.21	221.82	102.07	-4.96	-4.65	39.12	5.17
Si <sub>2</sub> OSe	Electron	0.59	1.39	200.14	109.98	-3.21	-4.10	773.26	110.56
	Hole	0.4	0.38	200.14	109.98	-11.95	-3.96	191.16	$1.01 \times 10^3$



as shown in Fig. 5. For example, the band structure of Si<sub>2</sub>OSe around the VBM is flatter than that around the CBM as presented in Fig. 5(a). The flatter the band structure at the band edge (the larger the curvature radius), the smaller the second derivative  $\partial^2 E(k)/\partial k^2$ , leading to a large effective mass.

Table 3 reveals that electron mobility and hole mobility are both directionally anisotropic. The electron mobility  $\mu_{2D}$  along the *x* axis is much higher than that along the *y* axis for both studied Janus structures. The electron mobilities of Si<sub>2</sub>OS are calculated to be  $\mu_{2D}^y = 727.98 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and  $\mu_{2D}^x = 443.60 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , which are high enough for applications in nano-electronic devices.<sup>58</sup> The electron mobilities of Si<sub>2</sub>OS monolayer are comparable with obtained results for Janus Si<sub>2</sub>SSe  $\mu_{2D}^y = 897.66 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and  $\mu_{2D}^x = 627.13 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ .<sup>35</sup> The anisotropy in electron mobility is evident in the Si<sub>2</sub>OSe monolayer, where the electron mobility along *x* direction  $\mu_{2D}^x$  is about 7 times higher than  $\mu_{2D}^y$ . The calculated results for carrier mobility of Janus Si<sub>2</sub>OS and Si<sub>2</sub>OXSe monolayers are listed in Table 3. It should also be mentioned that the holes in Si<sub>2</sub>OXSe have high mobility, up to  $1000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  along the *y* axis, which is greater than the electron mobility.

## 4 Conclusions

Based on DFT calculations, we have reported the structural, mechanical, electronic, and transport features of 2D Janus structures Si<sub>2</sub>OX (*X* = S, Se, Te). Our results demonstrated that the Si<sub>2</sub>OS and Si<sub>2</sub>OSe monolayers are highly stable structures while Si<sub>2</sub>OTe monolayer is an unstable structure with large negative frequencies in its phonon spectrum. The auxetic effect has been found in the two investigated stable monolayers with large NPR values. Especially, the Si<sub>2</sub>OS structure possesses a very high NPR, up to  $-0.234$  in the *x*-axis. The studied monolayers Si<sub>2</sub>OS and Si<sub>2</sub>OSe are both indirect band gap semiconductors and the transitions from indirect to direct band gap have been found in these structures when the biaxial strains are introduced. In addition, carrier mobility in both considered Janus structures is very high and anisotropic. Our findings reveal the novel properties of Janus Si<sub>2</sub>OX structures and great potential in multifunctional applications.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

N. A. P acknowledges the support from the Belarusian Republican Foundation for Fundamental Research (Grant No. F23RNF-049) and the Belarusian National Research Program "Convergence-2025".

## References

- C. W. Huang, W. Ren, V. C. Nguyen, Z. Chen, J. Wang, T. Sritharan and L. Chen, *Adv. Mater.*, 2012, **24**, 4170–4174.
- R. Lakes, *Science*, 1987, **235**, 1038–1040.

- X. Xu, Q. Zhang, M. Hao, Y. Hu, Z. Lin, L. Peng, T. Wang, X. Ren, C. Wang, Z. Zhao, C. Wan, H. Fei, L. Wang, J. Zhu, H. Sun, W. Chen, T. Du, B. Deng, G. J. Cheng, I. Shakir, C. Dames, T. S. Fisher, X. Zhang, H. Li, Y. Huang and X. Duan, *Science*, 2019, **363**, 723–727.
- F. Milstein and K. Huang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1979, **19**, 2030–2033.
- J. F. Silveira and A. R. Muniz, *Carbon*, 2017, **113**, 260–265.
- B. Saha and A. Datta, *J. Phys. Chem. C*, 2018, **122**, 19204–19211.
- B. Saha, S. M. Pratik and A. Datta, *Chem.–Eur. J.*, 2017, **23**, 12917–12923.
- J.-W. Jiang and H. S. Park, *Nat. Commun.*, 2014, **5**, 4727.
- A. J. Mannix, X.-F. Zhou, B. Kiraly, J. D. Wood, D. Alducin, B. D. Myers, X. Liu, B. L. Fisher, U. Santiago, J. R. Guest, M. J. Yacaman, A. Ponce, A. R. Oganov, M. C. Hersam and N. P. Guisinger, *Science*, 2015, **350**, 1513–1516.
- Y. Wang, F. Li, Y. Li and Z. Chen, *Nat. Commun.*, 2016, **7**, 11488.
- S. Woo, H. C. Park and Y.-W. Son, *Phys. Rev. B*, 2016, **93**, 075420.
- L. Yu, Q. Yan and A. Ruzsinszky, *Nat. Commun.*, 2017, **8**, 15224.
- X. Li, X. Qiang, Z. Gong, Y. Zhang, P. Gong and L. Chen, *Research*, 2021, **2021**, 1904839.
- A. Alderson and K. L. Alderson, *Proc. Inst. Mech. Eng.*, 2007, **221**, 565–575.
- W. Yang, Z.-M. Li, W. Shi, B.-H. Xie and M.-B. Yang, *J. Mater. Sci.*, 2004, **39**, 3269–3279.
- C. Huang and L. Chen, *Adv. Mater.*, 2016, **28**, 8079–8096.
- F. Scarpa, *IEEE Signal Process. Mag.*, 2008, **25**, 126–128.
- M. Avellaneda and P. J. Swart, *J. Acoust. Soc. Am.*, 1998, **103**, 1449–1467.
- T. C. Phong and L. T. T. Phuong, *Phys. B*, 2023, **666**, 415119.
- T. C. Phong, V. T. T. Vi and L. T. T. Phuong, *Phys. Lett. A*, 2023, **480**, 128946.
- T. C. Phong, N. T. Nam and L. T. T. Phuong, *Phys. Lett. A*, 2023, **474**, 128830.
- J.-H. Yang, Y. Zhang, W.-J. Yin, X. G. Gong, B. I. Yakobson and S.-H. Wei, *Nano Lett.*, 2016, **16**, 1110–1117.
- T. Jing, D. Liang, M. Deng and S. Cai, *J. Mater. Chem. C*, 2020, **8**, 10382–10389.
- A.-Y. Lu, H. Zhu, J. Xiao, C.-P. Chuu, Y. Han, M.-H. Chiu, C.-C. Cheng, C.-W. Yang, K.-H. Wei, Y. Yang, Y. Wang, D. Sokaras, D. Nordlund, P. Yang, D. A. Muller, M.-Y. Chou, X. Zhang and L.-J. Li, *Nat. Nanotechnol.*, 2017, **12**, 744.
- J. Zhang, S. Jia, I. Kholmanov, L. Dong, D. Er, W. Chen, H. Guo, Z. Jin, V. B. Shenoy, L. Shi and J. Lou, *ACS Nano*, 2017, **11**, 8192–8198.
- T. V. Vu, V. T. T. Vi, H. V. Phuc, A. I. Kartamyshev and N. N. Hieu, *Phys. Rev. B*, 2021, **104**, 115410.
- T. V. Vu, C. V. Nguyen, H. V. Phuc, A. A. Lavrentyev, O. Y. Khyzhun, N. V. Hieu, M. M. Obeid, D. P. Rai, H. D. Tong and N. N. Hieu, *Phys. Rev. B*, 2021, **103**, 085422.
- N. N. Hieu, H. V. Phuc, A. I. Kartamyshev and T. V. Vu, *Phys. Rev. B*, 2022, **105**, 075402.





- 29 T. V. Vu, H. V. Phuc, A. I. Kartamyshev and N. N. Hieu, *Appl. Phys. Lett.*, 2023, **122**, 061601.
- 30 T. V. Vu, V. T. T. Vi, C. V. Nguyen, H. V. Phuc and N. N. Hieu, *J. Phys. D: Appl. Phys.*, 2020, **53**, 455302.
- 31 T. V. Vu, V. T. T. Vi, H. V. Phuc, C. V. Nguyen, N. A. Poklonski, C. A. Duque, D. P. Rai, B. D. Hoi and N. N. Hieu, *J. Phys.: Condens. Matter*, 2021, **33**, 225503.
- 32 L. Dong, J. Lou and V. B. Shenoy, *ACS Nano*, 2017, **11**, 8242–8248.
- 33 A. Mogulkoc, Y. Mogulkoc, S. Jahangirov and E. Durgun, *J. Phys. Chem. C*, 2019, **123**, 29922–29931.
- 34 X. Yan, X. Cui, B. Wang, H. Yan, Y. Cai and Q. Ke, *Iscience*, 2023, **26**, 106731.
- 35 N. T. Hiep, C. Q. Nguyen and N. N. Hieu, *Appl. Phys. Lett.*, 2023, **123**, 092102.
- 36 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 37 G. Kresse and J. Furthmüller, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 38 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- 39 J. Neugebauer and M. Scheffler, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **46**, 16067–16080.
- 40 H. J. Monkhorst and J. D. Pack, *Phys. Rev. B: Solid State*, 1976, **13**, 5188–5192.
- 41 J. Heyd, G. E. Scuseria and M. Ernzerhof, *J. Chem. Phys.*, 2003, **118**, 8207.
- 42 S. Grimme, J. Antony, S. Ehrlich and H. Krieg, *J. Chem. Phys.*, 2010, **132**, 154104.
- 43 A. Togo, L. Chaput and I. Tanaka, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 094306.
- 44 S. Nosé, *J. Chem. Phys.*, 1984, **81**, 511.
- 45 J. Bardeen and W. Shockley, *Phys. Rev.*, 1950, **80**, 72.
- 46 K.-A. N. Duerloo, M. T. Ong and E. J. Reed, *J. Phys. Chem. Lett.*, 2012, **3**, 2871–2876.
- 47 X. Zhou, N. Zhou, C. Li, H. Song, Q. Zhang, X. Hu, L. Gan, H. Li, J. Lü, J. Luo, J. Xiong and T. Zhai, *2D Mater.*, 2017, **4**, 025048.
- 48 T. Jing, D. Liang, M. Deng and S. Cai, *J. Mater. Chem. C*, 2020, **8**, 10382–10389.
- 49 M. Born and K. Huang, *Am. J. Phys.*, 1955, **23**, 474.
- 50 N. T. Hung, A. R. T. Nugraha and R. Saito, *J. Phys. D: Appl. Phys.*, 2018, **51**, 075306.
- 51 P. Xiang, S. Sharma, Z. M. Wang, J. Wu and U. Schwingenschlögl, *ACS Appl. Mater. Interfaces*, 2020, **12**, 30731.
- 52 E. Cadelano, P. L. Palla, S. Giordano and L. Colombo, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 235414.
- 53 R. C. Cooper, C. Lee, C. A. Marianetti, X. Wei, J. Hone and J. W. Kysar, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 035423.
- 54 S.-D. Guo, *Phys. Chem. Chem. Phys.*, 2018, **20**, 7236–7242.
- 55 R. Peng, Y. Ma, Z. He, B. Huang, L. Kou and Y. Dai, *Nano Lett.*, 2019, **19**, 1227–1233.
- 56 J.-H. Yang, Q. Yuan, H. Deng, S.-H. Wei and B. I. Yakobson, *J. Phys. Chem. C*, 2017, **121**, 123–128.
- 57 W. Wan, S. Zhao, Y. Ge and Y. Liu, *J. Phys.: Condens. Matter*, 2019, **31**, 435501.
- 58 B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti and A. Kis, *Nat. Nanotechnol.*, 2011, **6**, 147–150.

