

Cite this: *Chem. Sci.*, 2024, 15, 4981

All publication charges for this article have been paid for by the Royal Society of Chemistry

Received 9th January 2024
Accepted 24th February 2024

DOI: 10.1039/d4sc00165f

rsc.li/chemical-science

Fe–Fe bonding in the rhombic Fe₄ cores of the Zintl clusters [Fe₄E₁₈]^{4−} (E = Sn and Pb)[†]

Wei-Xing Chen,^{‡a} Zi-Sheng Li,^{‡b} Harry W. T. Morgan,^{‡b} Cong-Cong Shu,^a Zhong-Ming Sun^{‡*a} and John E. McGrady^{‡*b}

We report here the synthesis and characterization of two endohedral Zintl-ion clusters, [Fe₄Sn₁₈]^{4−} and [Fe₄Pb₁₈]^{4−}, which contain rhombic Fe₄ cores. The Fe–Fe bond lengths are all below 2.5 Å, distinctly shorter than in the corresponding Cu clusters, indicating the presence of Fe–Fe bonding. Subtle differences in the structure of the Fe₄ core between the two clusters suggest that the change in tetrel element causes a change in electronic ground state, with a very short Fe–Fe bond length of 2.328 Å present across the diagonal of the rhombus in the lead case.

1 Introduction

Zintl clusters, and in particular those that contain encapsulated transition metals or f-elements, offer a fascinating insight into chemical bonding on the boundary between molecular systems and nanoparticles/nanoalloys.^{1–6} Some of the earliest examples, such as icosahedral [M@Pb₁₂]^{2−},^{7,8} are relatively simple from an electronic perspective because the endohedral metal has a closed-shell (d¹⁰) configuration, and the core-like nature of the d orbitals limits the degree of interaction with the cage. The incorporation of earlier transition metals with fewer d electrons raises the possibility of paramagnetism, as is observed in [Mn@Pb₁₂]^{3−} ((ref. 9) (a triplet)) and [Fe@Ge₁₀]^{3−} ((ref. 10) (a doublet)).^{11,12} The relative destabilisation of the d orbitals also leads to stronger interactions with the cage ('back-bonding') which can, ultimately, drive structural changes, from the deltahedra that are characteristic of the electron deficient regime, to 3-connected 'fullerene-like' structures such as that of [Ru@Ge₁₂]^{3−} that are the signature of more electron-rich cages.^{13–16}

The extension of these principles to larger clusters containing two or more open-shell transition metal ions opens up a further range of possibilities. If the encapsulated metal ions are dilute, cooperative magnetic phenomena (ferro- and anti-ferromagnetism) may emerge, while a closer approach brings covalent metal–metal bonding into play. In the latter scenario, metal–metal and metal–

cage bonding may be complementary: overlap between the metals will destabilise antibonding orbitals, potentially to the extent that electrons may be driven onto the cage. Opportunities to explore this landscape of possibilities have, up until now, been restricted by the relatively small number of well-characterized examples containing two or more endohedral metals. Amongst the very few Zintl-ion clusters with direct covalent metal–metal bonds, the Fe–Fe bond length in [Fe₂Ge₁₆]^{4−} (ref. 17) (Fig. 1(a)) is 2.636(3) Å whilst in [Fe₂–Sn₄Bi₈]^{3−},¹⁸ it is remarkably short, at 2.396(4) Å. Very recently, we have also reported the tri-iron cluster [Fe₃Sn₁₈]^{4−} (Fig. 1(b)) which contains a linear array of Fe centres, again with a very short Fe–Fe bond length of 2.4300(9) Å.¹⁹ These Zintl clusters are members of a wider family of iron clusters with direct covalent Fe–Fe bonds, which includes the Fe¹Fe¹ paddlewheel complex with ultra-short Fe–Fe bonds (2.127 Å), linear Fe₃ chains (Fig. 1(b))^{20,21} and triangular Fe₃ (ref. 22 and 23) and octahedral Fe₆ units.^{24–27} Ohki has also reported a series of hydride bridged Fe₄ and Fe₆ clusters based on a rhombic Fe₄ core, with Fe–Fe distances between 2.5 and 2.7 Å (Fig. 1(e)). Neidig and co-workers have also recently characterized the structure of the previously elusive 'Kochi cluster', a [Fe₈Me₁₂][−] (Fig. 1(f)), a distorted cube with a doublet ground state and averaged Fe–Fe bonds lengths of 2.43 Å.^{28,29} Paramagnetism is ubiquitous in the clusters shown in Fig. 1, and their electronic structure has typically been interpreted in terms of a delocalized manifold of Fe-based orbitals, occupied to maximise the multiplicity ('Hund's coupling').³⁰

In a previous paper we have reported the structures of the copper-containing clusters, [Cu₄Sn₁₈]^{4−} and [Cu₄Pb₁₈]^{4−}.³¹ We now extend that work to the earlier transition metals by reporting the properties of two new Zintl clusters, [Fe₄Sn₁₈]^{4−} and [Fe₄Pb₁₈]^{4−}, both of which contain rhombic Fe₄ cores, isostructural with the Cu analogues. The Fe–Fe separations (2.328 Å to 2.498 Å) are, however, shorter than those in [Cu₄E₁₈]^{4−} (2.52–2.55 Å), consistent with the presence of Fe–Fe bonding. The Fe₄ cores are superficially very similar, but subtle differences suggest that the nature of the Fe–Fe bonding may be

^aState Key Laboratory of Elemento-Organic Chemistry, Tianjin Key Lab for Rare Earth Materials and Applications, School of Materials Science and Engineering, Nankai University, Tianjin 300350, China. E-mail: sunlab@nankai.edu.cn

^bDepartment of Chemistry, University of Oxford, South Parks Road, Oxford OX1 3QR, UK. E-mail: john.mcgrady@chem.ox.ac.uk

[†] Electronic supplementary information (ESI) available: Further experimental and computational details, along with the optimized coordinates. CCDC 2206397, 2170117, 2101583. For ESI and crystallographic data in CIF or other electronic format see DOI: <https://doi.org/10.1039/d4sc00165f>

[‡] These authors contributed equally to this work.

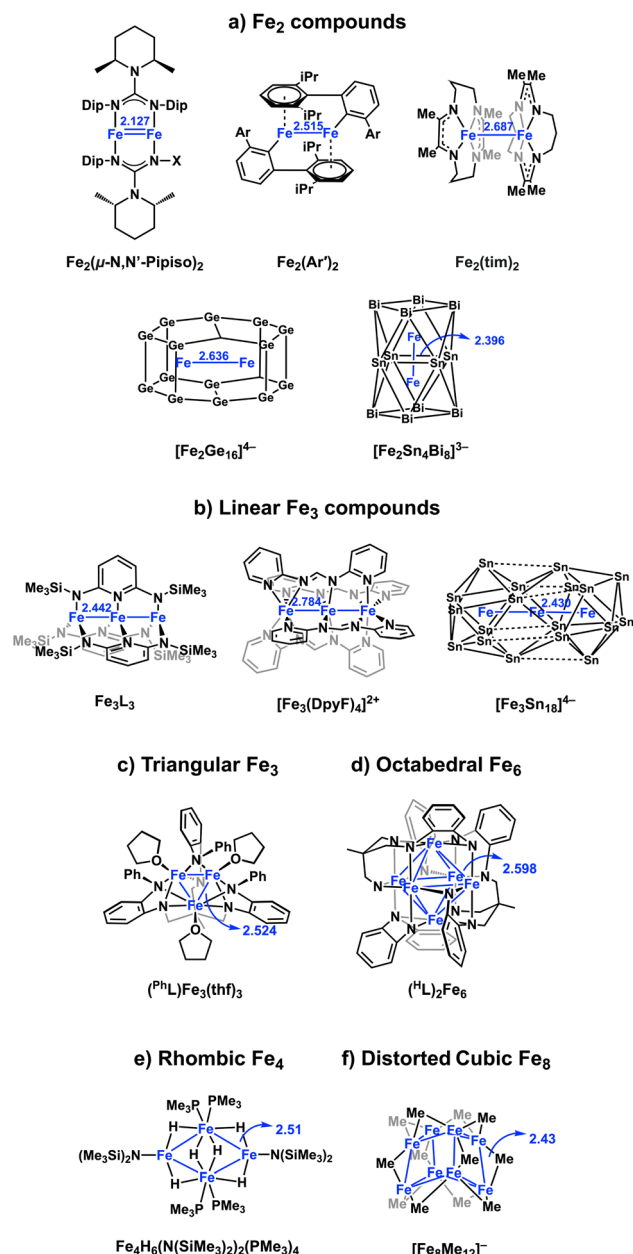


Fig. 1 Structures of selected Fe–Fe bonded clusters in the literature, including (a) Fe₂,^{17,32–34} (b) linear Fe₃,^{19–21} (c) triangular Fe₃,^{22,23} (d) octahedral Fe₆,^{24–26} (e) rhombic Fe₄,³⁵ (f) distorted cubic Fe₈.²⁸

different in the Sn₁₈ and Pb₁₈ clusters. This observation motivates a detailed survey of the electronic structure using density functional theory, where we draw on comparisons both between the Sn and Pb species and also between the Fe and Cu analogues.

2 Results and discussion

2.1 Synthesis and properties of [Fe₄E₁₈]^{4–}, E = Sn, Pb

[K(2.2.2-crypt)]₄[Fe₄Sn₁₈]·4Py (1) and [K(2.2.2-crypt)]₄[Fe₄Pb₁₈]·4Py (2), were obtained from the reactions of ethylenediamine solutions of K₄E₉ with ferrocene in the presence of 2.2.2-crypt

(further details are given in the experimental section). The reaction protocol differs from that used previously¹⁹ to synthesise the [Fe₃Sn₁₈]^{4–} cluster only in the source of Fe: ferrocene here but [K(thf)Fe(O^tBu)₃]₂ (thf = tetrahydrofuran). The two clusters, [Fe₃Sn₁₈]^{4–} and [Fe₄Sn₁₈]^{4–}, appear not to interconvert, even in the presence of excess Fe. 1 crystallizes in the monoclinic space group *P*₂₁/*n* and contains a single anionic stannaspherene unit [Fe₄Sn₁₈]^{4–} along with four [K(2.2.2-crypt)]⁺ cations and four pyridine molecules in the asymmetric unit (Fig. 2(a)). The [Fe₄Pb₁₈]^{4–} unit 2 is isostructural with 1 (Fig. 2(b)), both having approximate D_{2h} symmetry, as do their copper analogues, [Cu₄Sn₁₈]^{4–} and [Cu₄Pb₁₈]^{4–}.³¹ Two of the Fe atoms (Fe1 and Fe1') are encapsulated in 8-vertex E₈ polyhedra while Fe2 and Fe2' sit in approximately trigonal prismatic sites either side of the waist of the cluster. Averaged bond lengths for all four clusters are collected together in Table 1 for comparison. The gross features of the E₁₈ cage are largely unaffected by the change from Cu to Fe, with E–E bond lengths deviating by less than 0.05 Å in all cases. There are, however, subtle changes in M–M distances that reflect the emergence of metal–metal bonding in the iron clusters. As we have noted above, the d¹⁰ configuration of the Cu⁺ ions limits direct Cu–Cu bonding to the cuprophilic type, and so the short Cu1–Cu2 distances of ~2.53 Å must reflect, at least to some extent, the constraints imposed by the E₁₈ framework. Irrespective of whether the main-group element is Sn or Pb, the Fe1–Fe2 bond lengths are shorter than the Cu1–Cu2 analogues by approximately 0.1 Å. Differences in the M2–M2' distances are, in contrast, rather more variable: in the Sn clusters, the Fe2–Fe2' is shorter than Cu2–Cu2' by 0.135 Å, but this difference increases to 0.233 Å for the Pb pair. Indeed the Fe2–Fe2' distance of 2.328 Å in [Fe₄Pb₁₈]^{4–} is comparable to the shortest Fe–Fe bonds shown in Fig. 1. The correlation of bond length and bond strength is not always straightforward in clusters with bridging ligands, but we note that for a given oxidation state the radius of Fe is intrinsically larger than that of Cu, so the shorter Fe–Fe distances within a constant E₁₈ framework are a strong *a priori* indication of direct Fe–Fe bonding. Moreover, the wider variation in the M2–M2' bond lengths is probably a reflection of the fact that these atoms are not fully encapsulated by the cluster, and are therefore more free to move in response to subtle changes in Fe–Fe bonding. We explore the origins of these structural differences as part of a detailed analysis of electronic structure presented in the following sections.

Electrospray ionization mass spectra (ESI-MS) of freshly prepared samples of 1 and 2, measured in negative-ion mode, are shown in Fig. S9–15.† For 1, the spectrum shows an intense feature due to the dianion [Fe₄Sn₁₈]^{2–}, as well as smaller features due to the monoanion, [Fe₄Sn₁₈][–] and the dianion paired with a single [K(2.2.2-crypt)]⁺ cation, [K(2.2.2-crypt)Fe₄Sn₁₈][–]. The corresponding spectrum for 2, in contrast, shows a single dominant peak due to the decomposition product [FePb₁₂][–], most likely an endohedrally encapsulated icosahedron, similar in structure to the known [Mn@Pb₁₂]^{3–} cluster. The only possible trace of the intact cluster is a very small peak, just above the baseline, centred at *m/z* = 1977: it is clear that the parent cluster does not survive in any significant quantities



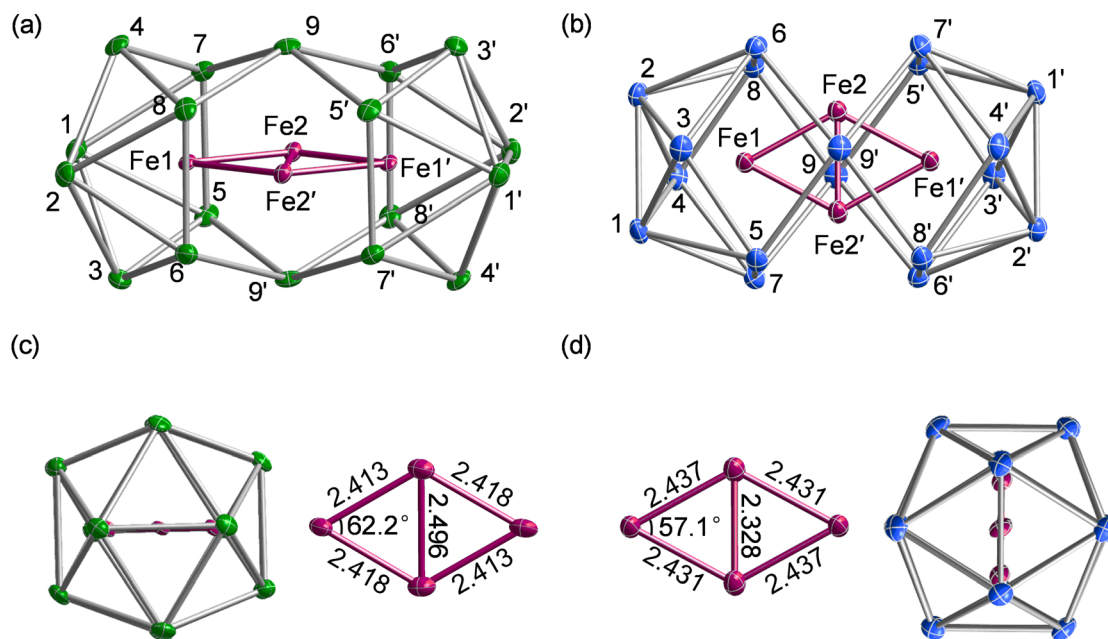


Fig. 2 Molecular structures of the anionic components, $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ and $[\text{Fe}_4\text{Pb}_{18}]^{4-}$. The two anions are isostructural: $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ is viewed along the x axis in (a) while $[\text{Fe}_4\text{Pb}_{18}]^{4-}$ is viewed along z in (b). Fe–Fe bond lengths in the Fe_4 rhombus are also shown in (c) and (d).

under ESI-MS conditions. The magnetic susceptibility of **1** was measured between 2 and 300 K using a superconducting quantum interference device (SQUID), Fig. 3. The measured magnetic moments (μ_{eff}) decreases with decreasing temperature from a value of $9.37 \mu_{\text{B}}$ ($\chi_{\text{M}}T = 10.97 \text{ cm}^3 \text{ K mol}^{-1}$) at 300 K to $\sim 9 \mu_{\text{B}}$ at 100 K, before dropping to $8.6 \mu_{\text{B}}$ at 10 K, and, finally, to much lower values below this temperature. The drop off below 10 K may be the result of zero-field splitting and/or weak antiferromagnetic coupling between cluster units. Limiting spin-only ($g = 2$) values of μ_{eff} for $S = 4$, $S = 5$ and $S = 6$ are $8.94 \mu_{\text{B}}$, $10.95 \mu_{\text{B}}$ and $12.96 \mu_{\text{B}}$, respectively, and the absence of a plateau at 300 K suggests that the ground spin state is not isolated. Despite multiple attempts on freshly prepared samples, we were not able to make reproducible magnetic

measurements on the lead analogue, **2**: this probably reflect the difficulties in separating crystals of **2** from other side products of the reaction while preparing the sample for magnetic measurements and/or rapid decomposition of the sample, as is apparent in the ESI-MS experiment.

2.2 Electronic structure

The electronic structure of the two Cu clusters, $[\text{Cu}_4\text{Sn}_{18}]^{4-}$ and $[\text{Cu}_4\text{Pb}_{18}]^{4-}$, was the subject of a previous study³¹ where we concluded that the ground states were singlets, with Cu^+ ions in a d^{10} configuration and no direct Cu–Cu interactions. In contrast, the open-shell Fe centres generate several closely-spaced electronic states that differ both in multiplicity and spatial symmetry, and the energies of the lowest-lying states

Table 1 Selected bond lengths from crystallographic and DFT-optimized structures (M06-L functional) for the $[\text{M}_4\text{E}_{18}]^{4-}$ family (all distances in Å)

		E_{rel}/eV	M1–M2	M2–M2'	M1–E3	M2–E9	E2–E6	E5–E9'	E7–E8'	Ref
$[\text{Fe}_4\text{Sn}_{18}]^{4-}$	X-ray (100 K)		2.413	2.496	2.828	2.739	3.142	3.225	4.898	This work
	DFT ($^9\text{A}_g$)	0.59	2.41	2.47	2.90	2.75	3.17	3.18	4.89	
	DFT ($^{11}\text{B}_{1g}$)	0.0	2.42	2.31	2.90	2.79	3.20	3.19	4.88	
	DFT ($^{11}\text{B}_{3g}$)	0.05	2.41	2.52	2.88	2.75	3.17	3.24	4.86	
	DFT ($^{13}\text{B}_{2g}$)	0.25	2.43	2.63	2.94	2.75	3.14	3.27	5.06	
$[\text{Fe}_4\text{Pb}_{18}]^{4-}$	X-ray (100 K)		2.437	2.328	2.974	2.892	3.247	3.287	4.941	This work
	DFT ($^9\text{B}_{1u}$)	0.79	2.41	2.46	2.99	2.90	3.32	3.37	4.99	
	DFT ($^{11}\text{B}_{1g}$)	0.0	2.44	2.28	3.03	2.90	3.35	3.32	4.99	
	DFT ($^{11}\text{B}_{3g}$)	0.18	2.43	2.48	3.00	2.86	3.32	3.36	4.98	
	DFT ($^{13}\text{B}_{3g}$)	0.18	2.46	2.35	3.01	2.86	3.34	3.37	5.05	
$[\text{Cu}_4\text{Sn}_{18}]^{4-}$	X-ray (100 K)		2.525	2.631	2.842	2.760	3.108	3.219	4.915	31
	DFT ($^1\text{A}_g$)		2.52	2.58	2.86	2.76	3.14	3.24	4.91	
$[\text{Cu}_4\text{Pb}_{18}]^{4-}$	X-ray (100 K)		2.547	2.561	2.935	2.825	3.224	3.299	5.012	31
	DFT ($^1\text{A}_g$)		2.55	2.52	2.99	2.88	3.30	3.35	5.02	

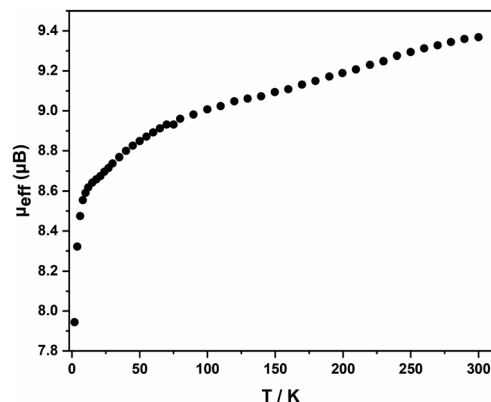


Fig. 3 Effective magnetic moment, μ_{eff} , as a function of temperature for 1.

with $2S + 1 = 9, 11$ and 13 , calculated with the M06-L functional, are summarized in Table 1. Amongst these, the two lowest share a common spin, $S = 5$, but have different spatial symmetries, $^{11}\text{B}_{1g}$ and $^{11}\text{B}_{3g}$: the lowest-lying state with $S = 4$ lies more than 0.5 eV higher in energy. The measured magnetic moment of $9.37 \mu_{\text{B}}$ for $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ at 300 K lies considerably below the spin-only value for $S = 5$ ($10.95 \mu_{\text{B}}$), and in fact is more consistent with a ground-state with $S = 4$. The apparent discrepancy between computed and experimental values may be a consequence of the well-known sensitivity of ground-state multiplicity to choice of density functional or, alternatively, may indicate the presence of diamagnetic impurities in the sample that serve to depress the measured moment. Of the two states with $2S + 1 = 11$, $^{11}\text{B}_{1g}$ is the most stable for both $[\text{Fe}_4\text{Pb}_{18}]^{4-}$ and $[\text{Fe}_4\text{Sn}_{18}]^{4-}$, although the energetic difference is too small to allow a confident assignment of the ground state, particularly in the Sn cluster where the two differ by only 0.05 eV. However, a comparison of optimized and measured bond lengths offers important clues to the identity of the electronic ground states. The optimised Fe2–Fe2' bond is short in the $^{11}\text{B}_{1g}$ state (2.31 Å and 2.28 Å for Sn and Pb, respectively), while the Fe1–Fe2 bond is relatively long (2.42 Å and 2.44 Å for Sn and Pb, respectively). In the $^{11}\text{B}_{3g}$ state, the opposite pattern is found: the Fe2–Fe2' bond is long (2.52 Å for Sn and 2.48 Å for Pb) while the Fe1–Fe2 bond is somewhat shorter (2.41 Å for Sn and 2.43 Å for Pb). When compared to experiment, it is clear that the $^{11}\text{B}_{3g}$ state offers a much better match to the X-ray structure of $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ (Fe1–Fe2 = 2.413 Å, Fe2–Fe2' = 2.496 Å) while the $^{11}\text{B}_{1g}$ state offers a better match to the X-ray structure of $[\text{Fe}_4\text{Pb}_{18}]^{4-}$ (Fe1–Fe2 = 2.437 Å, Fe2–Fe2' = 2.328 Å). On that basis, and despite the fact that the $^{11}\text{B}_{3g}$ state is 0.05 eV less stable than $^{11}\text{B}_{1g}$, we believe that former is the best candidate for the electronic ground state of the Sn cluster.

A schematic representation of the Kohn–Sham orbitals for $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ and $[\text{Fe}_4\text{Pb}_{18}]^{4-}$ in their respective $^{11}\text{B}_{3g}$ and $^{11}\text{B}_{1g}$ states is shown in Fig. 4, and a full analysis of the orbitals, projected densities of states (PDOS), and overlap-projected densities of states (OPDOS) is presented in the ESI.† The interpretation of the electronic structure is a complex challenge

due to the open-shell character, and a consideration from a symmetry-based perspective is a useful starting point. The rhombic Fe_4^{4+} unit has a total of $4 \times 5 = 20$ linear combinations of Fe 3d orbitals, which can be separated into in-plane ($a_g, b_{1g}, b_{2u}, b_{3u}$ symmetries) and out-of-plane (a_u, b_{1u}, b_{2g} and b_{3g}) subsets. The in-plane set support Fe–Fe interactions of local σ and π_{ip} symmetry, while the out-of-plane set have local Fe–Fe π_{op} and δ symmetry.‡ The formal oxidation state, and therefore the number of electrons occupying the Fe 3d manifold, is not straightforward due to the multiple redox states available to Fe. However, the structures of the E_{18} units in 1 and 2 and also the Sn/Pb densities of states (ESI, Fig. S22†) are strikingly similar to those in the $[\text{Cu}_4\text{E}_{18}]^{4-}$ cases, where the charge state of the E_{18} unit was assigned as 8^- .³¹ The relationship between structure and charge in the family of 18-vertex tetrel clusters has been discussed by us and others in the recent literature,^{19,36–40} and we provide further justification for the assignment of an 8^- charge state in the ESI.† Assuming the same 8^- charge for the E_{18} unit in $[\text{Fe}_4\text{E}_{18}]^{4-}$, we then have a net 4^+ charge for the Fe_4 unit and hence 28 valence electrons in the Fe 3d manifold. Distributing these 28 valence electrons across the 20 orbitals generates a maximum possible spin of $S = 6$ (the $^{13}\text{B}_{2g}$ state identified in Table 1). In the ground states of the two clusters with $2S + 1 = 11$, ($^{11}\text{B}_{1g}$ and $^{11}\text{B}_{3g}$) the highest of these 20 linear combinations, $7b_{1g}$, is vacant in both spin- α and spin- β manifolds, leaving 9 doubly-occupied and 10 singly-occupied orbitals. The doubly-occupied levels are largely Fe1–Fe2 and Fe2–Fe2' bonding while the singly-occupied levels are largely antibonding, giving rise to net Fe–Fe bonding, in contrast to the Cu analogues where all bonding and antibonding combinations are filled (see the discussion of the PDOS and OPDOS in the ESI† for more details). We can identify three distinct bonding/antibonding orbital pairs that make substantial contributions to the Fe2–Fe2' bonding in both the $^{11}\text{B}_{1g}$ and $^{11}\text{B}_{3g}$ states: the $8a_g^2 8b_{2u}^1$ configuration (Fe2–Fe2' σ), the $4b_{1u}^2 5b_{3g}^1$ configuration (π_{op}) and the $7b_{3u}^2 7b_{1g}^0$ configuration (π_{ip}). The key difference between the $^{11}\text{B}_{1g}$ and $^{11}\text{B}_{3g}$ states lies in the reversal of the $6b_{1g}$ and $4b_{3g}$ orbitals (highlighted in red and blue, respectively, in Fig. 4) the first of which has significant Fe2–Fe2' π_{ip} character while the second is localised largely on the Fe1 centres with little amplitude on the central Fe2–Fe2' unit. The in-plane component of the Fe2–Fe2' bond is therefore stronger in the $^{11}\text{B}_{1g}$ state, leading to compressed bond in $[\text{Fe}_4\text{Pb}_{18}]^{4-}$. The differences between Cu and Fe, and between $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ and $[\text{Fe}_4\text{Pb}_{18}]^{4-}$, are captured in the Mayer bond orders (MBO),⁴¹ delocalization indices (DI)⁴² and bond critical point ellipticities (ϵ) shown in the ESI, Table S6.† The Mayer bond orders of 0.51 and 0.91 for the Fe2–Fe2' bonds of $[\text{Fe}_4\text{Sn}_{18}]^{4-}$ and $[\text{Fe}_4\text{Pb}_{18}]^{4-}$ in their respective ground states are a clear indication of the stronger bond in the latter while the Fe1–Fe2 bonds show the opposite trend (0.85 and 0.69). The corresponding values are, on average, less than 0.2 for the Cu clusters, where formal Cu–Cu bond orders are zero. The ellipticity at the critical point, ϵ , provides a measure of the balance between σ , π_{ip} and π_{op} character: values close to 0 are indicative of a cylindrically symmetric environment while larger values indicate an anisotropic π component to the bond. The values of $\epsilon \sim 0.2$ for the Cu



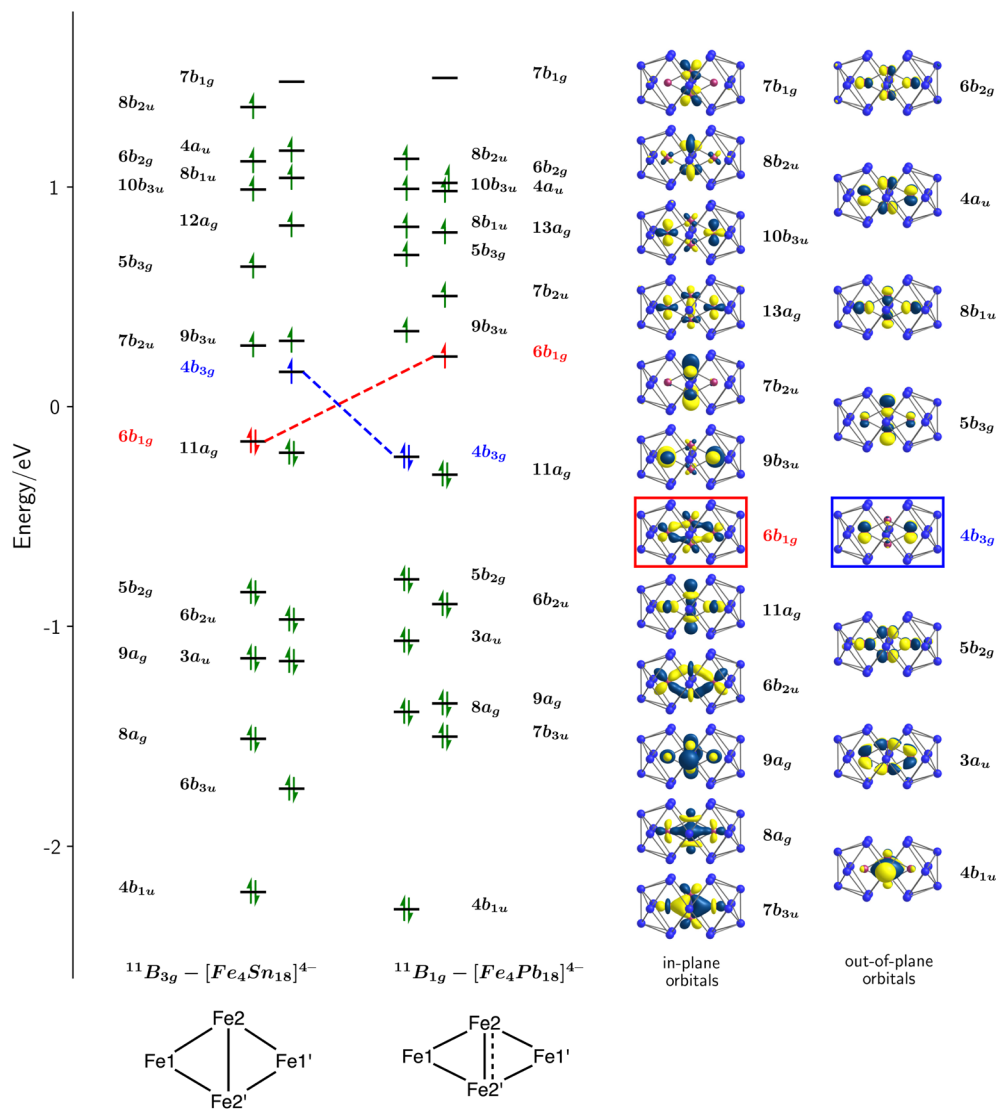


Fig. 4 Kohn-Sham molecular orbitals for the $^{11}B_{3g}$ of $[Fe_4Sn_{18}]^{4-}$ and the $^{11}B_{1g}$ state of $[Fe_4Pb_{18}]^{4-}$. The eigenvalues correspond to the spin- β manifold in both cases, and they are shifted such that $E = 0$ is defined as the mid-point between the eigenvalues of HOMO and LUMO. The orbitals highlighted in red and blue are those whose occupations differ in the $^{11}B_{3g}$ and $^{11}B_{1g}$ states. The isosurfaces shown correspond to the spin- β set of $[Fe_4Pb_{18}]^{4-}$.

clusters again establish a useful reference point. The ellipticity of 0.46 the Fe2-Fe2' bond of $[Fe_4Pb_{18}]^{4-}$ in its ground state is much lower than the value of 2.11 for the corresponding bond in the $^{11}B_{3g}$ ground state of $[Fe_4Sn_{18}]^{4-}$, the high value in the latter case reflecting the substantial contribution of the π_{op} bonding mediated by the $4b_{1u}/5b_{3g}$ pair. In the $^{11}B_{1g}$ ground state of $[Fe_4Pb_{18}]^{4-}$, this out-of-plane π component is supplemented by an additional π_{ip} component resulting from the removal of one electron from the Fe-Fe π antibonding $4b_{3g}$ orbital. In summary, our analysis of the electronic structure reveals that (a) There is significant Fe-Fe covalent bonding in both **1** and **2**, in contrast to the situation in the Cu analogues where bonding is absent and (b) $[Fe_4Sn_{18}]^{4-}$ and $[Fe_4Pb_{18}]^{4-}$ have different electronic configurations, leading to a marked change in the structure of the Fe₄ rhombus with a significantly contracted Fe2-Fe2' bond in the latter.

3 Summary and conclusions

In this paper we have reported the synthesis and characterization of two new Fe-based Zintl clusters, $[Fe_4Sn_{18}]^{4-}$, $[Fe_4Pb_{18}]^{4-}$, both of which have rhombic Fe₄ cores. The clusters are isostructural with the previously-reported Cu analogues, but the Fe-Fe bonds, all of which are below 2.5 Å, are substantially shorter than the corresponding Cu-Cu distances, indicating the presence of Fe-Fe bonding. There are also significant differences between the Fe₄ cores in the Sn and Pb clusters, with the Fe2-Fe2' bond across the diagonal of the rhombus being notably shorter in the latter (2.328 Å vs. 2.498 Å). Magnetic susceptibility measurements for $[Fe_4Sn_{18}]^{4-}$ indicate a high-spin ground state, with limiting values of μ_{eff} consistent with either $S = 5$ or $S = 6$. A survey of the potential energy surface using density functional theory identifies two almost

degenerate states with $S = 5$ as the most stable (M06-L functional), with very different Fe2–Fe2' bond lengths that map onto the differences seen in the crystallography. On that basis we believe that the two clusters adopt ground states with the same multiplicity but different spatial symmetry, with Fe–Fe π bonding more developed in the $[\text{Fe}_4\text{Pb}_{18}]^{4-}$ case.

4 Materials and methods

4.1 Materials and reagents

All manipulations and reactions were performed under a nitrogen atmosphere using standard Schlenk or glovebox techniques. Ethylenediamine (en) (Aldrich, 99%) and DMF (Aldrich, 99.8%) were freshly distilled by CaH_2 prior to use, and stored in N_2 prior to use. Tol (Aldrich, 99.8%) was distilled from sodium/benzophenone under nitrogen and stored under nitrogen. 2.2.2-crypt (4,7,13,16,21,24-Hexaoxa-1,10-diazabicyclo (8.8.8) hexacosane, purchased from Sigma-Aldrich, 98%) was dried in vacuum for one day prior to use. FeCp_2 was purchased from Energy Chemical (China). K_4E_9 was synthesized by heating a stoichiometric mixture of the elements (K: +99%, Sn: 99.99% and Pb: 99.99% all from Aladdin) at 850 °C for 36 h in a niobium tube.

4.2 Synthesis of 1 and 2

4.2.1 $[\text{K}(2.2.2\text{-crypt})]_4[\text{Fe}_4\text{Sn}_{18}] \cdot 4\text{Py}(1)$. In a 10 mL vial, K_4Sn_9 (122 mg, 0.1 mmol) and 2.2.2-crypt (115 mg, 0.3 mmol) were dissolved in en (*ca.* 3 mL) and stirred for 30 min, resulting a dark brown solution. Then FeCp_2 (30 mg, 0.161 mmol) was dispersed in toluene (0.5 mL), producing a light pink suspension, and then added dropwise to the above mixture. The mixture was stirred for 3 h at room temperature yielding a brown solution. All volatiles were removed at 60 °C under vacuum to obtain a black solid which was dissolved in 3 mL pyridine. The resulting green-black solution was stirred for 2 h at room temperature and filtered with glass wool. The filtrate was layered with 3 mL toluene. Block-shaped crystals of **1** were obtained (20% yield based on the used precursor K_4Sn_9) after six weeks.

4.2.2 $[\text{K}(2.2.2\text{-crypt})]_4[\text{Fe}_4\text{Pb}_{18}] \cdot 4\text{Py}(2)$. In a 10 mL vial, K_4Pb_9 (120 mg, 0.059 mmol), 2.2.2-crypt (88 mg, 0.234 mmol) and FeCp_2 (22 mg, 0.118 mmol) were dissolved in 3 mL en yielding a black solution. The mixture was stirred for 3 h at room temperature yielding a brown solution. All volatiles were removed at 60 °C under vacuum to obtain a dark-brown solid which was dissolved in 3 mL pyridine. The resulting green-black solution was stirred for 1 h at room temperature and filtered with glass wool. The filtrate was layered with 3 mL toluene. Black rod-like crystals of **2** were obtained (13% yield based on the used precursor K_4Pb_9) after one week.

4.3 Experimental analysis

4.3.1 X-ray diffraction. Suitable single crystals were selected for X-ray diffraction analyses. Crystallographic data were collected on Rigaku XtalAB Pro MM007 DW diffractometer with graphite monochromated Cu $K\alpha$ radiation ($\lambda = 1.54184 \text{ \AA}$).

4.3.2 Electrospray ionization mass spectrometry (ESI-MS) investigations. Negative ion mode ESI-MS of the DMF solutions of the single crystal and reaction solution were measured on an LTQ linear ion trap spectrometer by Agilent Technologies ESI-TOF-MS (6230). The spray voltage was 5.48 kV and the capillary temperature was kept at 300 °C. The capillary voltage was 30 V. The samples were made up inside a glovebox under a nitrogen atmosphere and rapidly transferred to the spectrometer in an airtight syringe by direct infusion with a Harvard syringe pump at 0.2 mL min^{-1} .

4.3.3 Energy dispersive X-ray (EDX) spectroscopic analysis. EDX analysis on the title clusters were performed using a scanning electron microscope (FE-SEM, JEOL JSM-7800F, Japan). Data acquisition was performed with an acceleration voltage of 15 kV and an accumulation time of 60 s.

4.3.4 Superconducting quantum interference devices (SQUID). Magnetic measurements were performed using a Quantum Design SQUID MPMS-3 magnetometer in the 2.0 to 300 K temperature range with an applied field of 1000 Oe.

4.4 Computational details

All calculations are performed using density functional theory as implemented in the ADF 2021.104 package.⁴³ The local functional of Minnesota 06 family (M06-L) functionals^{44,45} was used and triple-zeta basis sets included with two polarization functions are used for all atoms.⁴⁶ All electrons are treated as valence in the calculations. Relativistic effect was considered with The Zeroth-Order Regular Approximation (ZORA).⁴⁷ The numerical quality was set to 'verygood'.⁴⁸ A Conductor-like Screening Model (COSMO) with dielectric constant of 78.39 was used to simulate the confining environment of the ionic lattice.⁴⁹ Optimized structures were confirmed to be minima through the absence of imaginary frequencies.⁵⁰ The sensitivity of the results to choice of functional was also explored by repeating the calculations using the PBE and PBE0 functionals.^{51,52}

Data availability

Yes, the original data that supports the finding in the article are available in the ESI† and CCDC databases.

Author contributions

W.-X. C. and C.-C. S. performed the synthesis and characterisation, Z. L. and H. W. T. M. performed the computational analysis. Z.-M. S. and J. E. M. conceived the project and supervised the experimental and computational aspects of the research, respectively. All authors contributed to the preparation of the manuscript.

Conflicts of interest

There are no conflicts to declare.



Acknowledgements

This work was supported by the National Natural Science Foundation of China (no. 92161102 and 22371140), the Natural Science Foundation of Tianjin City (no. 21JCZJC00140) and the PhD Candidate Research Innovation Fund of NKU School of Materials Science and Engineering, and 111 project (B18030) from China (MOE).

Notes and references

§ ip and op denote in-plane and out-of-plane, respectively.

- 1 J. Zhao, Q. Du, S. Zhou and V. Kumar, *Chem. Rev.*, 2020, **120**, 9021–9163.
- 2 B. Weinert and S. Dehnen, in *Binary and Ternary Intermetallic Clusters*, ed. S. Dehnen, Springer International Publishing, Cham, 2017, pp. 99–134.
- 3 B. Oelkers, M. V. Butovskii and R. Kempe, *Chem. – An Euro. J.*, 2012, **18**, 13566–13579.
- 4 N. Lichtenberger, R. J. Wilson, A. R. Eulenstein, W. Massa, R. Clérac, F. Weigend and S. Dehnen, *J. Am. Chem. Soc.*, 2016, **138**, 9033–9036.
- 5 X. Min, I. A. Popov, F.-X. Pan, L.-J. Li, E. Matito, Z.-M. Sun, L.-S. Wang and A. I. Boldyrev, *Angew. Chem., Int. Ed.*, 2016, **55**, 5531–5535.
- 6 L.-F. Cui, X. Huang, L.-M. Wang, J. Li and L.-S. Wang, *Angew. Chem., Int. Ed.*, 2007, **46**, 742–745.
- 7 E. N. Esenturk, J. Fettingner and B. Eichhorn, *J. Am. Chem. Soc.*, 2006, **128**, 9178–9186.
- 8 L.-F. Cui and L.-S. Wang, *Int. Rev. Phys. Chem.*, 2008, **27**, 139–166.
- 9 B. Zhou, T. Krämer, A. L. Thompson, J. E. McGrady and J. M. Goicoechea, *Inorg. Chem.*, 2011, **50**, 8028–8037.
- 10 B. Zhou, M. S. Denning, D. L. Kays and J. M. Goicoechea, *J. Am. Chem. Soc.*, 2009, **131**, 2802–2803.
- 11 X. Jin and J. E. McGrady, *Adv. Inorg. Chem.*, 2019, **73**, 265–304.
- 12 K. Mayer, J. Weßing, T. F. Fässler and R. A. Fischer, *Angew. Chem., Int. Ed. Engl.*, 2018, **57**, 14372–14393.
- 13 G. Espinoza-Quintero, J. C. A. Duckworth, W. K. Myers, J. E. McGrady and J. M. Goicoechea, *J. Am. Chem. Soc.*, 2014, **136**, 1210–1213.
- 14 J. E. McGrady, F. Weigend and S. Dehnen, *Chem. Soc. Rev.*, 2022, **51**, 628–649.
- 15 J. M. Goicoechea and J. E. McGrady, *Dalton Trans.*, 2015, **44**, 6755–6766.
- 16 X. Jin, V. Arcisauskaite and J. E. McGrady, *Dalton Trans.*, 2017, **46**, 11636–11644.
- 17 H. W. T. Morgan, K.-S. Csizi, Y.-S. Huang, Z.-M. Sun and J. E. McGrady, *J. Phys. Chem. A*, 2021, **125**, 4578–4588.
- 18 Y.-N. Yang, Z.-S. Li, S. Mondal, L. Qiao, C.-C. Wang, W.-J. Tian, Z.-M. Sun and J. E. McGrady, *Chin. Chem. Lett.*, 2023, 109048.
- 19 Z.-S. Li, W.-X. Chen, H. W. T. Morgan, C.-C. Shu, J. E. McGrady and Z.-M. Sun, *Chem. Sci.*, 2024, **15**, 1018–1026.
- 20 A. Srinivasan, R. A. Musgrave, M. Rouzières, R. Clérac, J. E. McGrady and E. A. Hillard, *Chem. Commun.*, 2021, **57**, 13357–13360.
- 21 G. L. Guillet, K. Y. Arpin, A. M. Boltin, J. B. Gordon, J. A. Rave and P. C. Hillesheim, *Inorg. Chem.*, 2020, **59**, 11238–11243.
- 22 E. V. Eames, T. D. Harris and T. A. Betley, *Chem. Sci.*, 2012, **3**, 407–415.
- 23 R. Hernández Sánchez, A. K. Bartholomew, T. M. Powers, G. Ménard and T. A. Betley, *J. Am. Chem. Soc.*, 2016, **138**, 2235–2243.
- 24 Q. Zhao, T. D. Harris and T. A. Betley, *J. Am. Chem. Soc.*, 2011, **133**, 8293–8306.
- 25 R. Hernández Sánchez and T. A. Betley, *J. Am. Chem. Soc.*, 2015, **137**, 13949–13956.
- 26 R. H. Sánchez and T. A. Betley, *J. Am. Chem. Soc.*, 2018, **140**, 16792–16806.
- 27 J. Nehrkorn, S. M. Greer, B. J. Malbrecht, K. J. Anderton, A. Aliabadi, J. Krzystek, A. Schnegg, K. Holldack, C. Herrmann, T. A. Betley, S. Stoll and S. Hill, *Inorg. Chem.*, 2021, **60**, 4610–4622.
- 28 S. B. I. Muñoz, S. L. Daifuku, W. W. Brennessel and M. L. Neidig, *J. Am. Chem. Soc.*, 2016, **138**, 7492–7495.
- 29 C. L. Kwan and J. K. Kochi, *J. Am. Chem. Soc.*, 1976, **98**, 4903–4912.
- 30 J. A. Chipman and J. F. Berry, *Chem. Rev.*, 2020, **120**, 2409–2447.
- 31 L. Qiao, C. Zhang, C.-C. Shu, H. W. T. Morgan, J. E. McGrady and Z.-M. Sun, *J. Am. Chem. Soc.*, 2020, **142**, 13288–13293.
- 32 C. Hess, T. Weyhermüller, E. Bill and K. Wieghardt, *Angew. Chem., Int. Ed. Engl.*, 2009, **48**, 3703–3706.
- 33 T. Nguyen, W. Merrill, C. Ni, H. Lei, J. Fettingner, B. Ellis, G. Long, M. Brynda and P. Power, *Angew. Chem., Int. Ed. Engl.*, 2008, **47**, 9115–9117.
- 34 L. Fohlmeister, S. Liu, C. Schulten, B. Moubaraki, A. Stasch, J. D. Cashion, K. S. Murray, L. Gagliardi and C. Jones, *Angew. Chem., Int. Ed.*, 2012, **51**, 8294–8298.
- 35 R. Araake, K. Sakadani, M. Tada, Y. Sakai and Y. Ohki, *J. Am. Chem. Soc.*, 2017, **139**, 5596–5606.
- 36 F. K. Sheong, W.-J. Chen, J.-X. Zhang, Y. Li and Z. Lin, *Dalton Trans.*, 2017, **46**, 2214–2219.
- 37 J.-X. Zhang, F. K. Sheong and Z. Lin, *Inorg. Chem.*, 2019, **58**, 3473–3478.
- 38 J.-X. Zhang, F. K. Sheong and Z. Lin, *Wiley Interdiscip. Rev. Comput. Mol. Sci.*, 2020, **10**, e1469.
- 39 F. S. Geitner, W. Klein and T. F. Fässler, *Dalton Trans.*, 2017, **46**, 5796–5800.
- 40 J.-Q. Wang, B. Wahl and T. Fässler, *Angew. Chem., Int. Ed.*, 2010, **49**, 6592–6595.
- 41 I. Mayer, *Chem. Phys. Lett.*, 1983, **97**, 270–274.
- 42 J. Poater, M. Solà, M. Duran and X. Fradera, *Theor. Chem. Acc.*, 2002, **107**, 362–371.
- 43 G. te Velde, F. M. Bickelhaupt, E. J. Baerends, C. Fonseca Guerra, S. J. A. van Gisbergen, J. G. Snijders and T. Ziegler, *J. Comput. Chem.*, 2001, **22**, 931–967.
- 44 Y. Zhao and D. G. Truhlar, *J. Chem. Phys.*, 2006, **125**, 194101.
- 45 Y. Zhao and D. G. Truhlar, *Theor. Chem. Acc.*, 2008, **120**, 215–241.



- 46 E. Van Lenthe and E.-J. Baerends, *J. Comput. Chem.*, 2003, **24**, 1142–1156.
- 47 E. van Lenthe, A. Ehlers and E.-J. Baerends, *J. Chem. Phys.*, 1999, **110**, 8943–8953.
- 48 M. Franchini, P. H. T. Philipsen and L. Visscher, *J. Comput. Chem.*, 2013, **34**, 1819–1827.
- 49 C. C. Pye and T. Ziegler, *Theor. Chem. Acc.*, 1999, **101**, 396–408.
- 50 L. Versluis and T. Ziegler, *J. Chem. Phys.*, 1988, **88**, 322–328.
- 51 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 52 C. Adamo and V. Barone, *J. Chem. Phys.*, 1999, **110**, 6158–6170.

