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# A-site cation manipulation of exemplary second harmonic generation response and optical anisotropy in rare-earth borates†

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Ultraviolet nonlinear optical (UV NLO) materials have garnered significant interest for their prospective applications in advanced laser technologies. However, tailoring the desired structure in these materials remains a formidable challenge. Here, we propose a simple yet effective strategy for synthesizing rare-earth borates,  $K_xNa_{3-x}La_2B_3O_9$  ( $x = 2-3$ ), by manipulating the A-site cations to induce structural evolution. Notably,  $K_xNa_{3-x}La_2B_3O_9$  undergoes a phase transition from the *Pnc2* to the *Amm2* space group by adjusting the  $K^+$  content to reach  $x = 2.6$ . Moreover, the target compounds exhibit strong phase-matching second harmonic generation (SHG) efficiencies, ranging from 1.3 to 3.3 times that of KDP ( $KH_2PO_4$ ), and feature short UV cutoff edges of around 204–208 nm. Additionally, the correlation between microscopic polarizability, optical anisotropy, and the structural evolution of these materials was characterized through structural and theoretical analyses. These findings highlight the potential applications of  $K_xNa_{3-x}La_2B_3O_9$  as UV NLO materials and underscore the viability of manipulating A-site cations to fabricate NLO crystals with desirable properties.

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## Introduction

Nonlinear optical (NLO) crystals have garnered considerable attention in scientific and industrial applications due to their ability to manipulate laser frequency and produce coherent light. In particular, ultraviolet (UV) NLO materials have emerged as crucial contenders for extending the wavelength range of solid-state lasers into the UV region, facilitated by a second harmonic generation (SHG) process.<sup>1–3</sup> These materials hold significant promise in various optoelectronic fields such as laser processing, short-wave communication, semiconductor lithography, biomedicine, and high-density storage.<sup>4–7</sup> Despite considerable efforts and advancements in exploring acentric materials, the practical realization of UV NLO crystals remains limited. Noteworthy examples include  $KH_2PO_4$  (KDP),  $\beta-BaB_2O_4$  ( $\beta$ -BBO),  $LiB_3O_5$  (LBO),  $KBe_2BO_3F_2$  (KBBF), and  $CsLiB_6O_{10}$  (CLBO), among others.<sup>8–12</sup> In response to the development of laser technology, there is an increasing demand for novel NLO crystals capable of directly generating UV coherent

light. However, the fabrication of NLO crystals faces challenges stemming from the inherent propensity of inorganic materials to exhibit undesired dipole–dipole interactions or spatial effects, tending to crystallize centrosymmetric structures.<sup>13,14</sup> Consequently, the development of innovative synthetic strategies to exploit NLO crystals with SHG-enhanced activity remains a significant challenge.

Borates, characterized by their versatile structural configurations and favorable UV transparency attributed to strong covalent B–O bonds, have emerged as a candidate system for exploring UV optical crystals.<sup>15–19</sup> To effectively innovate and construct novel NLO crystals, the introduction of highly polarizable chromophores into borates is an optional strategy to modify the structure and enhance the optical properties. Strategies such as incorporating second-order Jahn–Teller (SOJT) active cations ( $Zr^{4+}$ ,  $Nb^{5+}$ ,  $Ta^{4+}$ , etc.), stereo-chemically active lone-pair cations ( $Te^{4+}$ ,  $Bi^{3+}$ , etc.), or  $d^{10}$  transition metal cations ( $Zn^{2+}$ ,  $Cd^{2+}$ , etc.) have been widely recognized in the quest for new optical materials.<sup>20–22</sup> Following these design strategies, numerous borates with acentric structures have been synthesized, including  $BaZr(BO_3)_2$ ,  $CsNbB_2O_6$ ,  $K_3M_3B_2O_{12}$  ( $M = Nb, Ta$ ),  $\alpha-BiB_3O_6$ ,  $CaBi_2B_2O_7$ ,  $Cd_4BiO(BO_3)_3$ ,  $Bi_3TeBO_9$ ,  $Na_2ZnB_6O_{11}$ ,  $Cs_3Zn_6B_9O_{21}$ , and  $CaZn_2(BO_3)_2$ .<sup>2,23–31</sup> However, the strategies for excavating UV NLO crystals are still limited due to challenges in accurately regulating the structure.

Recent studies have elucidated the indispensable role of alkali metal cations (A-site cations) in the exploration of novel NLO materials, due to the fact that alkali metal cations are

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devoid of d-d or f-f electron transitions, enabling the broadening of the bandgap and consequently achieving short absorption edges.<sup>32</sup> Additionally, alkali metal cations with various ionic radii also have a significant impact on the spatial molecular arrangements and macroscopic symmetry. Notable examples include  $\text{RbB}_4\text{O}_6\text{F}$  ( $0.8 \times \text{KDP}$ ),  $\text{CsB}_4\text{O}_6\text{F}$  ( $1.9 \times \text{KDP}$ ),  $\text{CsRbB}_8\text{O}_{12}\text{F}_2$  ( $1.1 \times \text{KDP}$ ),  $\text{K}_7\text{SrY}_2\text{B}_{15}\text{O}_{30}$  ( $1.1 \times \text{KDP}$ ),  $\text{Rb}_7\text{SrY}_2(\text{B}_5\text{O}_{10})_3$  ( $0.9 \times \text{KDP}$ ),  $\text{Na}_3\text{La}_2(\text{BO}_3)_3$  ( $2 \times \text{KDP}$ ), and  $\text{KNa}_2\text{La}_2(\text{BO}_3)_3$  ( $2.6 \times \text{KDP}$ ),<sup>33–38</sup> exhibiting different NLO activities attributed to the presence of diverse A-site cations. A similar situation has also been observed in alkali metal borophosphate and phosphate UV NLO crystals, such as  $\text{Rb}_3\text{B}_{11}\text{P}_2\text{O}_{23}$  ( $2.5 \times \text{KDP}$ ),  $\text{Cs}_3\text{B}_{11}\text{P}_2\text{O}_{23}$  ( $3 \times \text{KDP}$ ),  $\alpha\text{-KZnPO}_4$  ( $0.2 \times \text{KDP}$ ) and  $\alpha\text{-LiZnPO}_4$  ( $2.3 \times \text{KDP}$ ).<sup>39–41</sup> These investigations suggest that the optimization of local structures induced by A-site cations may enhance SHG performance. However, limited research has been conducted thus far on the evolution process of the structure and performance of distinct A-site cations in these NLO borate-based derivatives.

To tackle this challenge, we utilized rare earth borates as a foundation and introduced alkali metals with varying ionic radii to optimize the structural arrangement and optical properties by adjusting the composition of alkali metals. For instance, the substitution of  $\text{Na}^+$  and  $\text{K}^+$  cations is widely regarded as a viable approach due to the inherent flexibility in doping between these components.<sup>42</sup> Additionally, the incorporation of rare earth cations, like  $\text{Sc}^{3+}$ ,  $\text{Y}^{3+}$ ,  $\text{La}^{3+}$ ,  $\text{Gd}^{3+}$  and  $\text{Lu}^{3+}$ , not only contributes to excellent transparency in the UV region, but also offers various coordination types.<sup>43</sup> Notably, the  $\text{La}^{3+}$  cation with a sizable ionic radius tends to form flexible SHG-active chromophores, thereby generating favorable SHG effects.<sup>38</sup> Inspired by these ideas, our work focuses on rare earth borate-based materials and explores the underlying mechanism of the NLO effect and optical anisotropy by employing a facile

strategy to manipulate the components of A-site alkali metal cations. The synthesis of rare-earth borates  $\text{K}_x\text{Na}_{3-x}\text{La}_2\text{B}_3\text{O}_9$  ( $x = 2-3$ ) was achieved through structural evolutions attributed to A-site cation manipulations. As anticipated, these compounds displayed remarkable phase-matching SHG efficiencies ranging from 1.3 to 3.3 times that of KDP, while featuring short UV cutoff edges of approximately 204 nm. These findings highlight the potential of  $\text{K}_x\text{Na}_{3-x}\text{La}_2\text{B}_3\text{O}_9$  as favorable candidates for UV NLO applications and underscore the significance of regulating A-site alkali metal cations in the exploration of new NLO crystals.

## Results and discussion

### Synthesis and phase transformation

The preparation of the target compounds was performed through a modified solid-phase reaction. To monitor the influence of A-site cations on the crystal structure, a series of compounds with varying proportions of A-site cations, namely  $\text{K}_2$ ,  $\text{K}_{2.2}$ ,  $\text{K}_{2.4}$ ,  $\text{K}_{2.6}$ ,  $\text{K}_{2.8}$ , and  $\text{K}_3$ , were synthesized. The phase purity of the resulting products was verified using PXRD analysis. It is evident from Fig. 1a that the  $\text{K}_2$  and  $\text{K}_3$  compounds demonstrate distinct crystal structural characteristics. Specifically, the structural characteristics of  $\text{K}_{2.8}$  and  $\text{K}_3$  exhibited isomorphism, suggesting a consistent structural framework despite the varying proportions of K cations in the A-site. However, when the  $\text{K}^+$  cation proportion reached approximately  $\text{K}_{2.6}$ , structural features indicative of the  $\text{K}_2$  compound were initially observed, implying the coexistence of two distinct polycrystalline phases. This observation confirms that the structural transition from  $\text{K}_3$  to  $\text{K}_2$  occurs at around the  $\text{K}_{2.6}$  proportion. Furthermore, the PXRD peak patterns of  $\text{K}_2$ ,  $\text{K}_{2.2}$ , and  $\text{K}_{2.4}$  displayed a high degree of similarity, suggesting isomorphism between these compounds. These

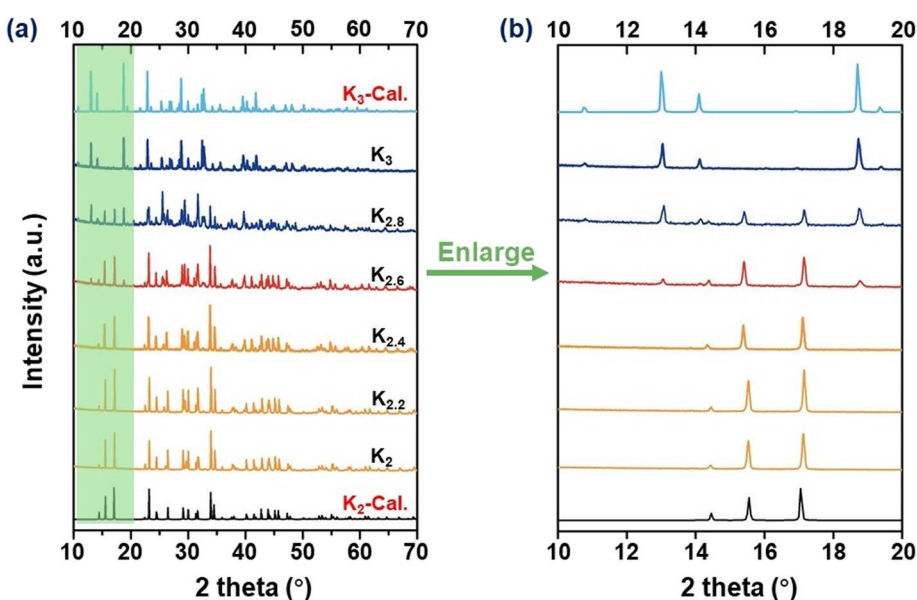


Fig. 1 (a) Experimental and calculated PXRD patterns for  $\text{K}_2$ ,  $\text{K}_{2.2}$ ,  $\text{K}_{2.4}$ ,  $\text{K}_{2.6}$ ,  $\text{K}_{2.8}$ , and  $\text{K}_3$  polycrystals, respectively. (b) The enlarged PXRD curves observed in the  $2\theta$  range of  $10-20^\circ$ .

observations provide clear evidence of the critical impact of the A-site cation proportions on the structural regulation of these materials. Notably, as illustrated in Fig. 1b, an increase in the K proportion within the range of  $K_2$ – $K_{2.4}$  leads to a gradual shift of PXRD peaks towards lower angles, indicating larger lattice constants.<sup>44</sup> This trend is also observed when considering the proportion of K variation within the  $K_{2.8}$ – $K_3$  range. These findings strongly suggest that the proportion of  $K^+$  cations exerts a direct impact on the lattice constants of these compounds.

### Thermal analyses

Thermal analyses were conducted to further investigate the structural transformations and evaluate the thermal stability of the title compounds. TG-DTA measurements revealed the remarkable thermal stability of the  $K_2$ ,  $K_{2.2}$ ,  $K_{2.4}$ ,  $K_{2.6}$ ,  $K_{2.8}$ , and  $K_3$  compounds, as evidenced by sharp absorption peaks observed above 850 °C (Fig. 2a). Notably, the thermal stability of compounds within the  $K_2$ – $K_{2.6}$  range exhibited a decreasing trend with increasing proportions of  $K^+$  cations (Fig. 2b). In contrast, within the  $K_{2.6}$ – $K_3$  range, the compounds displayed an upward trend in thermal stability (Fig. 2g). Among the tested compound systems, it has been observed that the  $K_{2.6}$  compound demonstrated the lowest thermal stability. When the  $K^+$  cation reaches a composition of 2.6, this behavior can potentially be attributed to the influence of mixed components present in the system that cannot be ignored, leading to a decrease in the melting point of the system. These findings further highlight the role of the  $K_{2.6}$  compound, which possesses mixed-phase structures that align with the PXRD analyses.

### Crystal structures of $K_2$ and $K_3$

In light of the isomorphism observed among the  $K_2$ ,  $K_{2.2}$ , and  $K_{2.4}$  compounds, as well as the isomorphism between the  $K_{2.8}$  and  $K_3$  compounds, we have selected  $K_2$  and  $K_3$  as representative samples for conducting a thorough analysis of their respective crystal structures. The structural features of  $K_2$  and  $K_3$  were determined through the utilization of single-crystal XRD analysis (Tables S1–S5†). Our analysis revealed that  $K_2$  adopts an asymmetric orthorhombic crystal structure, belonging to the space group *Amm*2 (no. 38), while  $K_3$  exhibits a similar orthorhombic structure in the *Pnc*2 space group (no. 30), as summarized in Table S1.† In the crystal structure of  $K_2$ , the asymmetric unit consists of two distinct K, one Na, one La, two B, and four O atoms. Conversely, the asymmetric unit of  $K_3$  contains three K, three La, four B, and eleven O atoms. As shown in Fig. 3a and b, the B atoms are coordinated by three O atoms, forming  $[BO_3]$  plane triangles, and the La atom is surrounded by nine O atoms, resulting in the formation of distorted  $[LaO_9]$  polyhedra. The B–O bond lengths in  $K_2$  range from 1.360(2) to 1.379(12) Å, while the La–O bond lengths range from 2.466(12) to 2.687(5) Å. In  $K_3$ , the B–O bond lengths vary from 1.300(5) to 1.409(15) Å, whereas the La–O bond lengths range from 2.412(8) to 3.02(2) Å. In  $K_2$ , adjacent  $[LaO_9]$  polyhedra are connected to each other through oxygen edge-sharing, forming a pseudo-one-dimensional (1D) chain denoted as  $[La_2O_{16}]_\infty$ . These 1D chains, along with  $[BO_3]$  units, further interconnect through oxygen corner-sharing to form pseudo-two-dimensional (2D) layers on the *bc* plane (Fig. 3c). These pseudo-2D layers are then bridged with  $[BO_3]$  units along the *a*-axis through oxygen corner-sharing, ultimately giving rise to a three-dimensional (3D) structural framework (Fig. 3d). In the case of  $K_3$ , three distorted  $[LaO_9]$

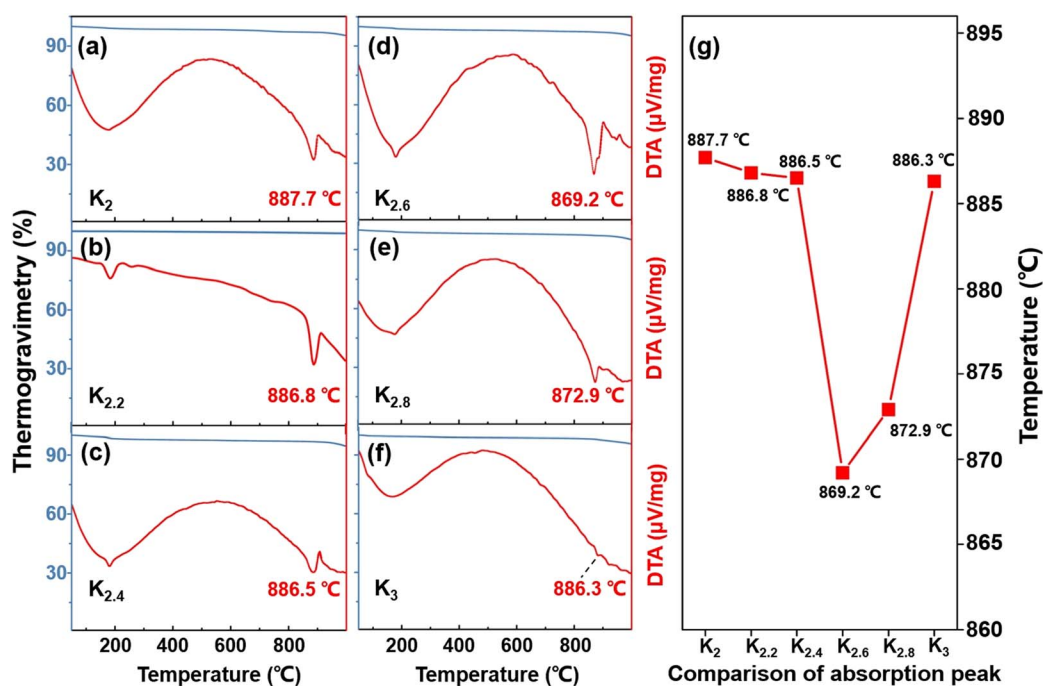


Fig. 2 (a–f) TG-DTA curves for the  $K_2$ ,  $K_{2.2}$ ,  $K_{2.4}$ ,  $K_{2.6}$ ,  $K_{2.8}$ , and  $K_3$  compounds, respectively. (g) Comparative results of thermal stabilities for the title compounds.



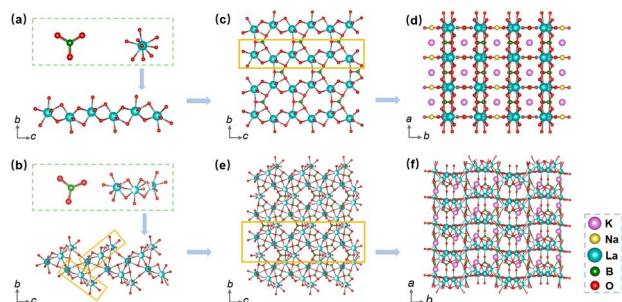


Fig. 3 Structural characteristics of  $K_2$  and  $K_3$ . (a and b)  $[BO_3]$  plane triangles,  $[LaO_9]$  polyhedra, and  $[La_2O_{16}]_\infty$  and  $[La_3O_{21}]$  clusters. (c) Pseudo-2D layers of  $K_2$ . (d) 3D structural framework of  $K_2$ . (e) Presentation of  $[La_3O_{21}]_\infty$  clusters connected with  $[BO_3]$  units in  $K_3$  viewed from the  $bc$  plane. (f) 3D structural network of  $K_3$  viewed along the  $c$ -axis.

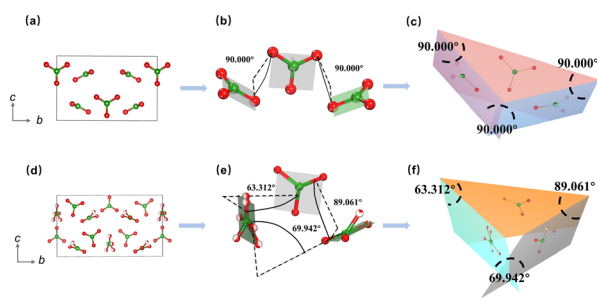


Fig. 4 Comparison of the arrangements of distinct  $[BO_3]$  units in the  $K_2$  and  $K_3$  compounds. (a–c) Arrangements and orientations of the isolated  $[BO_3]$  units in the  $K_2$  compound. (d–f) Arrangements and orientations of the isolated  $[BO_3]$  units in the  $K_3$  compound.

polyhedra associate to form distinct  $[La_3O_{21}]$  clusters *via* oxygen face-sharing. These clusters are further interconnected with  $[BO_3]$  units by sharing oxygen along the  $bc$  plane, as presented in Fig. 3e, with the  $K^+$  cations located in the channels (Fig. 3f).

To shed light on the structural evolution exhibited by the  $K_2$  and  $K_3$  compounds, our study focuses on a thorough analysis of the arrangement and interrelationships of fundamental  $[BO_3]$  units. Notably, while both  $K_2$  and  $K_3$  compounds feature isolated planar triangles formed by  $[BO_3]$  units, a notable disparity arises in the spatial arrangement of these triangles within their respective 3D frameworks. As shown in Fig. 4a–c, it is noteworthy that the two distinct  $[BO_3]$  plane triangles in  $K_2$  exhibit a regular arrangement, manifesting a constant angle of 90 degrees between the planes they occupy. In contrast, the planes defined by the distinct  $[BO_3]$  units in  $K_3$  feature varying angles, including 63.312, 89.061, and 69.942 degrees, which deviate significantly from the uniform 90 degree arrangement observed in  $K_2$ . The variations in the spatial angles between the  $[BO_3]$  building blocks in  $K_2$  and  $K_3$  have distinct implications for the anisotropic characteristics of their respective structures. These observations would contribute to a more comprehensive understanding of the potential structural and application-related implications associated with the different spatial arrangements of  $[BO_3]$  units in  $K_2$  and  $K_3$ .

### Spectroscopic properties

To investigate the optical properties of the title compounds, the UV-vis-NIR diffuse reflectance spectra of the title compounds were recorded under consistent conditions (Fig. 5). The obtained spectra revealed distinct UV transmittance cut-off edges for each compound: 208, 206, 206, 207, 208, and 204 nm for the  $K_2$ ,  $K_{2.2}$ ,  $K_{2.4}$ ,  $K_{2.6}$ ,  $K_{2.8}$ , and  $K_3$  compounds, respectively. These favorable UV transmittance characteristics directly correspond to energy band gaps of 5.15, 5.24, 5.20, 5.19, 5.23, and 5.29 eV, respectively, as determined using the Kubelka–Munk formula.<sup>45</sup> Furthermore, we conducted a detailed analysis of the infrared (IR) spectra of the compounds, which revealed similar characteristics among them, as depicted in Fig. S1.† Taking the  $K_2$  compound as an example, the absorption peaks observed at around 1392.73 and 1200.69  $cm^{-1}$  are attributed to the asymmetric stretching

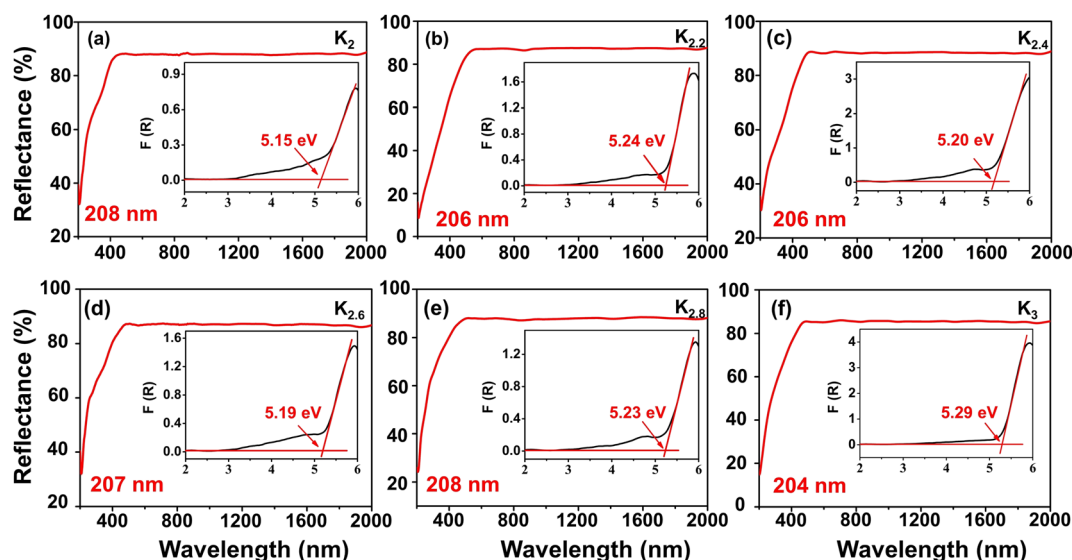


Fig. 5 (a–f) UV-vis-NIR diffuse reflectance spectra for the  $K_2$ ,  $K_{2.2}$ ,  $K_{2.4}$ ,  $K_{2.6}$ ,  $K_{2.8}$ , and  $K_3$  polycrystalline samples, respectively.





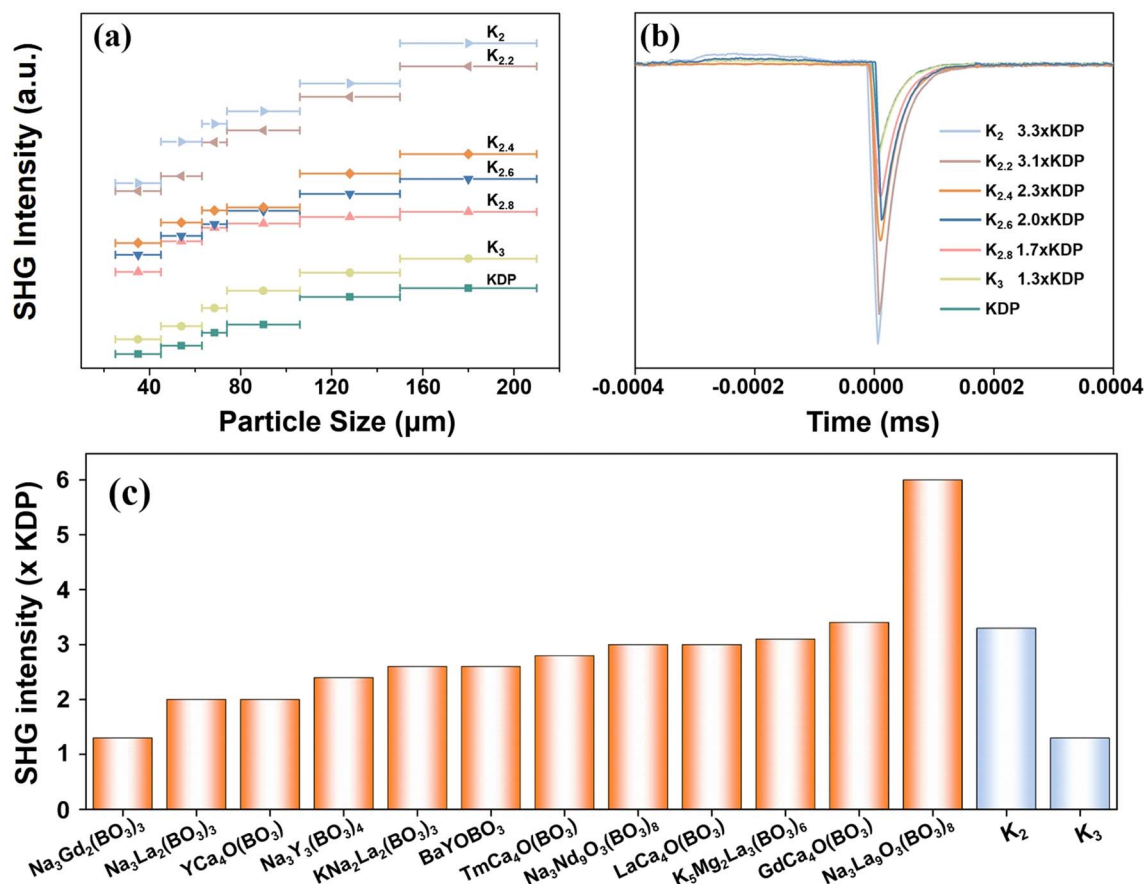


Fig. 6 (a) Phase-matched curves for KDP, K<sub>2</sub>, K<sub>2.2</sub>, K<sub>2.4</sub>, K<sub>2.6</sub>, K<sub>2.8</sub>, and K<sub>3</sub> polycrystalline powders under 1064 nm laser irradiation. (b) Comparison of SHG intensity between the KDP reference and the title compounds. (c) Comparison of the reported SHG intensity of alkali/alkaline earth metal rare earth borates containing [BO<sub>3</sub>] units.

vibrations of the [BO<sub>3</sub>] units, while the peak near 950.08 cm<sup>-1</sup> corresponds to the symmetric stretching vibrations. The peaks around 743.24 and 604.83 cm<sup>-1</sup> are caused by the bending vibrations of the [BO<sub>3</sub>] units. The presence of these distinct vibration modes further confirms the existence of the [BO<sub>3</sub>] unit within the compounds.<sup>37,38,43b</sup> Notably, we observed a slight redshift phenomenon in the absorption peak positions of the IR spectrum with an increasing K to Na component ratio. This trend can primarily be attributed to the expansion of the alkali metal cation radius.<sup>46</sup> With a larger cation radius, the bond length between the cation and oxygen atom increases, resulting in a lower vibration frequency and a consequent redshift in the absorption peak position of the IR absorption spectrum. These findings provide valuable insights into the intrinsic relationship between structural changes and the optical properties exhibited by the investigated compounds.

### SHG characterization

The title compounds are considered to possess SHG responses because they crystallize in non-centrosymmetric space groups. Fig. 6a presents a plot illustrating the correlation between the particle size and the detected powder SHG signals. It is observed that the SHG response intensity of each compound increases

with increasing particle size until it reaches its maximum value, suggesting phase-matching behavior in accordance with the rules proposed by Kurtz and Perry.<sup>47</sup>

Furthermore, the SHG intensities of K<sub>2</sub>, K<sub>2.2</sub>, K<sub>2.4</sub>, K<sub>2.6</sub>, K<sub>2.8</sub>, and K<sub>3</sub> were determined to be 3.3, 3.1, 2.3, 2.0, 1.7, and 1.3 times that of the benchmark KDP, respectively, within a particle size range of 177 to 210 μm (Fig. 6b). The SHG responses are sufficiently large for UV NLO applications. Notably, the significant disparity in the SHG effect between K<sub>2</sub> and K<sub>3</sub> can primarily be attributed to the disordered arrangement of oxygen atoms in K<sub>3</sub>, which gives rise to an inconsistent alignment of the [BO<sub>3</sub>] triangular units, consequently resulting in a lower microscopic second-order hyper-polarizability compared to K<sub>2</sub>.<sup>48</sup> Additionally, as the ratio of K to Na in these compounds gradually increases, the SHG response correspondingly decreases. This observation may be closely related to the unit cell volume of the crystal structure, where a larger unit cell volume tends to weaken the second-order polarizability. Fig. 6c and Table S6† provide a compilation of the reported SHG intensities of alkali and alkaline earth metal borate compounds containing [BO<sub>3</sub>] triangular units from the literature.<sup>49</sup> Notably, K<sub>2</sub> exhibits a significant SHG response compared with other reported compounds, thereby bolstering its potential as a NLO crystal.



## Birefringence characterization

The birefringence of  $K_2$  and  $K_3$  crystals was evaluated within the visible wavelength range using a polarizing microscope. Fig. 7a–d show the complete extinction of both  $K_2$  and  $K_3$  crystals under orthogonally polarized light. The measured crystal thicknesses ( $d$ ) were found to be 29  $\mu\text{m}$  and 16  $\mu\text{m}$ , corresponding to the optical path differences ( $R$ ) of 841 nm and 896 nm,<sup>50</sup> respectively. Consequently, the birefringence values of  $K_2$  and  $K_3$  crystals were determined to be 0.029 and 0.056, respectively, across the visible wavelength range. The theoretical birefringent values of the two crystals were also calculated, yielding values of 0.028 and 0.051 for  $K_2$  and  $K_3$  crystals (Fig. 7g and h), respectively, at a wavelength of 1  $\mu\text{m}$ , which aligns with the experimental observations. Notably, the birefringence value of  $K_3$  is significantly higher than that of  $K_2$ , nearly double that of  $K_2$ . This significant difference primarily stems from the angles between the  $[\text{BO}_3]$  planes in  $K_2$  and  $K_3$ , as discussed in the “Crystal structure” section. Specifically, the spatial structure of  $K_2$  consists of two distinct perpendicular  $[\text{BO}_3]$  units oriented at a 90 degree angle to each other, indicating relatively small structural anisotropy. In contrast,  $K_3$  displays an arrangement in which the  $[\text{BO}_3]$  units form varying angles, including 63.312, 89.061, and 69.942 degrees, thereby deviating from perpendicularity. This distinct structural feature in  $K_3$  results in relatively large structural anisotropy.<sup>29</sup>

## Structure–property correlations

To delve deeper into the electronic structures of  $K_2$  and  $K_3$  crystals and their inherent relationships with optical properties,

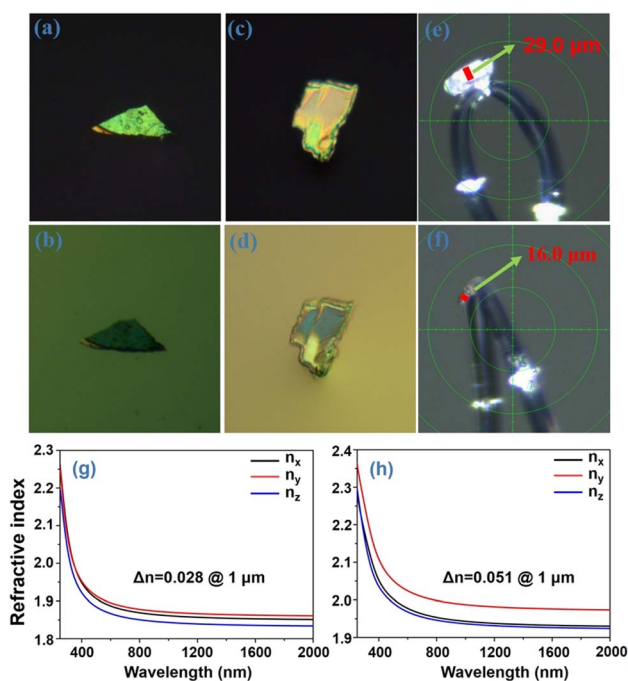


Fig. 7 (a–d) Original interference state and complete extinction of  $K_2$  and  $K_3$  crystals under cross-polarized light, respectively. (e and f) The thickness of the  $K_2$  and  $K_3$  crystals, respectively. (g and h) Theoretical dispersion curve of the refractive index for  $K_2$  and  $K_3$  crystals, respectively.

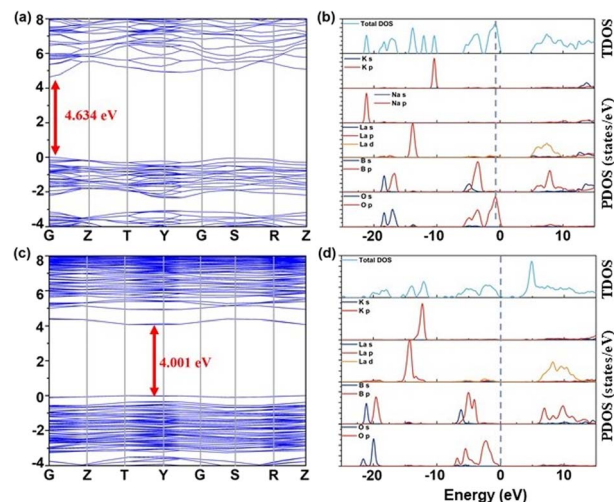


Fig. 8 (a and b) The presentation of the band structure as well as the PDOS and TDOS for  $K_2$ , respectively. (c and d) The presentation of the band structure as well as the PDOS and TDOS for  $K_3$ , respectively.

this study employed DFT-based computational methods for theoretical research.<sup>51</sup> Through first-principles calculations,<sup>52</sup> we discovered that both  $K_2$  and  $K_3$  crystals are direct bandgap compounds, indicating that the lowest point of the conduction band and the highest point of the valence band overlap in momentum space ( $k$ -space). The calculated theoretical bandgap values for  $K_2$  and  $K_3$  crystals are 4.634 eV and 4.001 eV, respectively, as shown in Fig. 8a–c. The orbital contributions of each atom to the energy bands of  $K_2$  and  $K_3$  can be identified from the corresponding partial densities of states (PDOSs). As shown in Fig. 8b–d, below the Fermi level, the valence band maximum predominantly consists of B 2p and O 2p orbitals, while the conduction band minimum primarily originates from the La 5d orbitals in both  $K_2$  and  $K_3$  compounds. It is noteworthy that the electron states in  $K_2$  predominantly occupied by K 3p orbitals range from  $-12$  to  $-11$  eV, whereas those in  $K_3$ , governed by K 3p orbitals, span from  $-13$  to  $-11$  eV. Given the close correlation between the optical properties and the optical transitions occurring between electronic states in proximity to the band gap, it can be deduced that the optical activities of the  $K_2$  and  $K_3$  crystals are primarily influenced by the  $[\text{BO}_3]$  units and  $[\text{LaO}_9]$  polyhedra.

## Conclusions

In summary, a new family of alkaline-metal rare-earth borates,  $K_x\text{Na}_{3-x}\text{La}_2\text{B}_3\text{O}_9$  ( $x = 2-3$ ), was synthesized through A-site cation control engineering. The investigation of these compounds *via* DTA-TG analyses and PXRD tests revealed a structural transition from the *Pnc2* to the *Amm2* space group. Notably, the structural phase of the  $K_x\text{Na}_{3-x}\text{La}_2\text{B}_3\text{O}_9$  compounds was found to be modulated by adjusting the A-site cation  $\text{K}^+$  content. Intriguingly, these materials display large SHG intensities, which are 1.3 to 3.3 times that of KDP with phase-matchable ability, along with short UV absorption edges at approximately 204–208 nm. Moreover, the  $\text{K}_2\text{NaLa}_2\text{B}_3\text{O}_9$  and

$K_3La_2B_3O_9$  crystals selected for analysis possess moderate birefringence values of 0.029 and 0.056, respectively, with the variation primarily attributed to differing orientations of the  $[BO_3]$  planes. These favorable findings make  $K_2NaLa_2B_3O_9$  and  $K_3La_2B_3O_9$  good potential candidates for UV NLO applications. The rational regulation of A-site cations underscores a versatile approach for the fabrication of novel NLO crystals with desired performance attributes.

## Data availability

The data supporting this article have been included as part of the ESI.† Crystallographic data for 2374500 and 2374501 have been deposited at the CCDC.

## Author contributions

Jie Song and Huijian Zhao: experiment, investigation, data curation, writing original draft. Conggang Li: conceptualization, funding acquisition, methodology, project administration, review & editing. Ning Ye, Zhanggui Hu and Yicheng Wu: resources, funding acquisition.

## Conflicts of interest

There are no conflicts to declare.

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