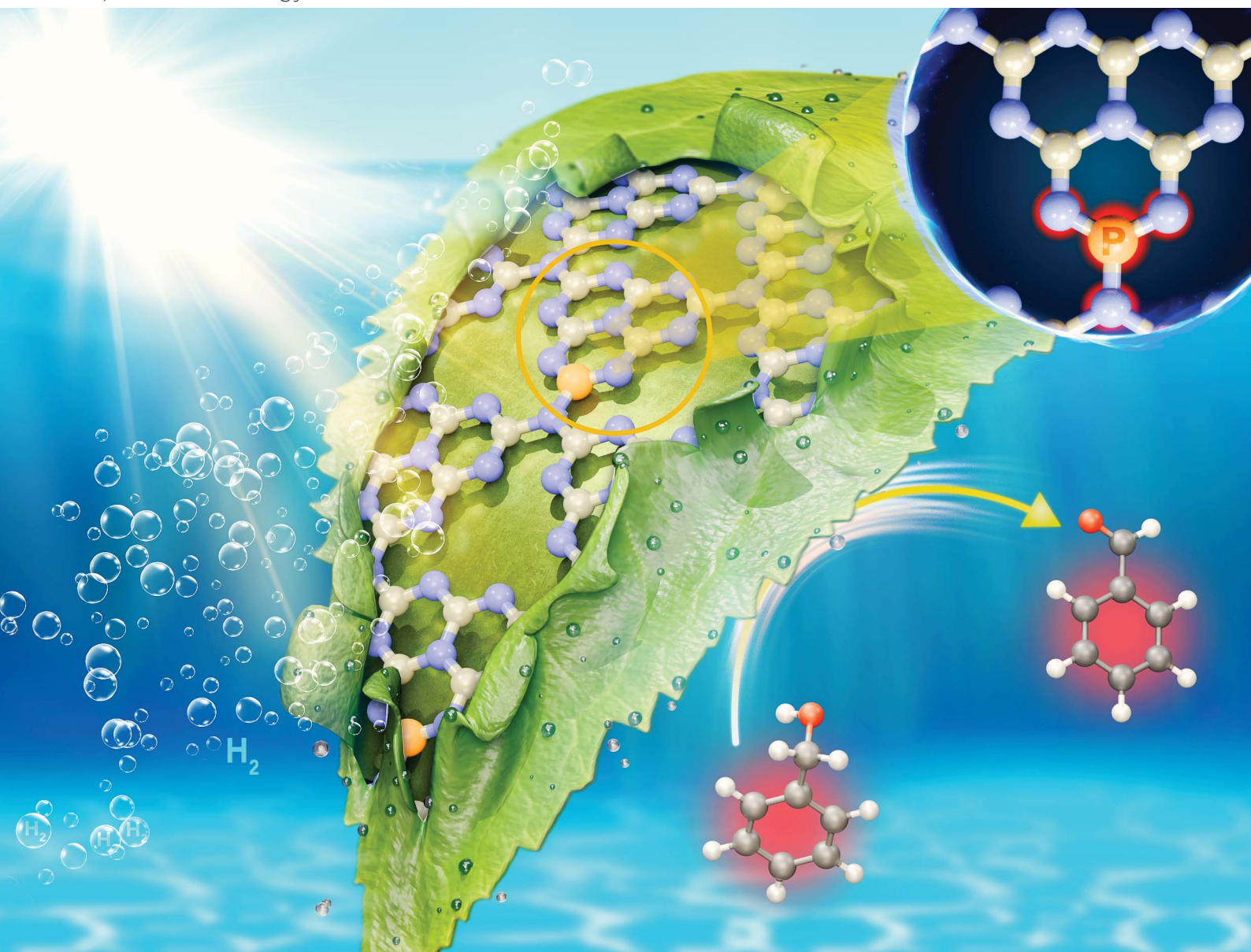


Sustainable Energy & Fuels

Interdisciplinary research for the development of sustainable energy technologies

rsc.li/sustainable-energy



ISSN 2398-4902

PAPER

Kiran Vankayala *et al.*

Boosted photoredox capability of visible light-active
P-doped C_3N_4 with efficient harvesting of electron-hole pairs

Cite this: *Sustainable Energy Fuels*,
2024, 8, 4914

Boosted photoredox capability of visible light-active P-doped C₃N₄ with efficient harvesting of electron–hole pairs†

Asmita Dileep Gaonkar,¹ Shradha Paniya,¹ Srinivasu Kancharlapalli^{1,2}
and Kiran Vankayala^{1*}

Photocatalytic production of solar fuels and high-value chemicals by photogenerated carriers has been at the forefront as one of the promising sustainable approaches. However, most of the studies focus only on one of the half reactions, either photoreduction or photooxidation, leading to underutilization of the potentiality of photocatalysis due to inefficient harvesting of electron–hole pairs. Herein, the efficient utilization of photogenerated electron–hole pairs was demonstrated by employing phosphorous-doped graphitic carbon nitride (P-doped g-C₃N₄) as a visible light-active photocatalyst that is capable of simultaneously producing hydrogen and benzaldehyde from benzyl alcohol. P-doping into g-C₃N₄ was achieved using an eco-friendly P source. P doping induced changes in the light-harvesting capacity of g-C₃N₄, and its consequence on the dual-functional photocatalytic activity of P-doped g-C₃N₄ was systematically investigated using various characterization techniques. P-doped g-C₃N₄ exhibited an ≈3-fold increase in photocatalytic activity in the production of H₂ and benzaldehyde as compared to that of pristine g-C₃N₄. Density functional theory (DFT) studies reveal that the P-dopant preferentially replaces the corner C-site as compared to the N site of the tri-s-triazine ring of g-C₃N₄, which results in the creation of mid-gap states that enable the enhanced visible light absorption of P-doped g-C₃N₄. Mechanistic investigation studies suggest that photogenerated holes drive the selective oxidation of benzyl alcohol to benzaldehyde while photogenerated electrons drive H₂ evolution, leading to concomitant production of H₂ and benzaldehyde by P-doped g-C₃N₄. The selective conversion of benzyl alcohol proceeds through a carbon-centred radical mechanism, according to experimental and DFT studies. This work elucidates the importance of P-doping in g-C₃N₄ for the simultaneous production of solar fuels (such as H₂) and high-value chemicals (such as benzaldehyde).

Received 21st June 2024
Accepted 17th August 2024

DOI: 10.1039/d4se00826j

rsc.li/sustainable-energy

Introduction

The global demand for energy is rising significantly, leading to the development of sustainable alternate energy sources due to the depletion of non-renewable fossil fuels.¹ The efficient conversion of abundant renewable solar energy into clean and storable chemical fuels has long been recognised as a viable and effective solution to the world's rising energy needs. In recent years, photocatalytic hydrogen (H₂) production *via* water splitting using visible light-active semiconductors has been considered a promising strategy for addressing the energy

crisis.^{2,3} In semiconductor-based photocatalytic H₂ evolution from pure water, photoexcited electrons reduce protons and generate H₂, whereas photogenerated holes oxidize water to evolve oxygen (O₂). The efficiency of photocatalytic H₂ evolution from pure water is severely constrained by the recombination of photoexcited electron–hole pairs, sluggish water-oxidation kinetics, and high-water oxidation overpotential.⁴ To increase the efficiency of H₂ generation *via* photocatalytic means, studies are often conducted in the presence of sacrificial electron donors (SEDs) such as lactic acid, triethanolamine (TEOA), and methanol. However, associated costs and waste production pose additional challenges, which may impede the practical implementation. Additionally, the oxidation power of photogenerated holes is under-utilized as they participate in unwanted oxidation reactions. Therefore, it would be beneficial if one uses accessible feedstock or feedstock-derived molecules as SEDs in order to improve H₂ evolution and profitability, as the oxidation of feedstock may generate high-value chemicals at the same time.^{5,6} Thus, benzyl alcohol (BA), a biomass-derived molecule,⁵ can be an interesting SED as the selective

¹Functional Materials for Electrochemistry and Solar Energy (FunMatES) group, Energy and Environmental Chemistry Lab, Department of Chemistry, Birla Institute of Technology and Science, Pilani, K. K. Birla Goa Campus, NH 17B Bypass Road, Zuarinagar, Goa 403726, India. E-mail: kiranv@goa.bits-pilani.ac.in; kiran2cu@gmail.com

²Chemistry Division, Bhabha Atomic Research Centre, Mumbai, India

³Homi Bhabha National Institute, Anushaktinagar, Mumbai, India

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4se00826j>



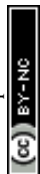
photocatalytic oxidation of BA yields benzaldehyde (BADH) in an eco-friendly manner as compared to traditional methods that involve partial oxidation of toluene or hydrolysis of benzylidene chloride.⁷ BADH is an important industrial solvent and a key intermediate for different perfumes, dyes, *etc.*⁷ The simultaneous production of H₂ and oxidation of BA to BADH using photogenerated electrons and holes, respectively, can be a novel approach to harvest holes and electrons effectively.⁸ In order to realize this, the selection of a suitable bifunctional semiconductor photocatalyst is an important step.⁸ Among the various photocatalysts, graphitic carbon nitride (g-C₃N₄) has gained immense interest due to its good thermal and chemical stability, visible-light response, and tunable electronic structure.⁹ In 2009, Wang *et al.* have first described the use of g-C₃N₄ as a promising visible-light-responsive photocatalyst for H₂ production that triggered the research on g-C₃N₄ for various photocatalytic applications.¹⁰ However, it should be noted that poor separation of charge carriers, and fast recombination rate provide opportunities to further explore this material.^{11,12} In recent years, various modification approaches have been implemented to enhance the photocatalytic activity of g-C₃N₄ such as thermal exfoliation of bulk g-C₃N₄ into two-dimensional (2D) nanosheets, constructing heterojunctions, heteroatom doping and band gap engineering.^{9,12–21} Non-metallic heteroatom doping-assisted tunable C₃N₄ has attracted immense attention to attain high activity by creating effective catalytic sites and band edges. The heteroatom is known to create localised states between the valence band (VB) and conduction band (CB) *via* atomic orbital hybridisation, which is beneficial to achieve wide visible absorption and to produce more active sites.¹³ Additionally, the dopants play a crucial role in lowering the recombination rates of charge carriers and thereby increasing the effective utilization of photocarriers that participate in surface reactions.¹⁴ Among various heteroatoms such as C, S, P, and O-doped g-C₃N₄, P-dopants have received considerable attention in the recent past because they may create P–N linkages that act as new migration channels for the charge carriers.¹⁵ Notably, it has been reported that the delocalized orbitals (HOMO and LUMO) around P and neighbouring N atoms help in achieving facile charge separation.^{14,15} For instance, Shi *et al.* reported the synthesis of P-doped C₃N₄ using hexachlorocyclotriphosphazene as a P precursor and reported 2.9-fold increment in H₂ evolution activity as compared to undoped C₃N₄.²² Qiao *et al.* have used 2-aminoethylphosphonic acid as a P-doping precursor to synthesize P-doped C₃N₄ nanosheets that exhibited 3.6 times higher photocatalytic H₂ evolution activity than that of bulk carbon nitride.¹⁶ Mu *et al.* have reported the synthesis of P-doped C₃N₄ for dye degradation study using 1-butyl-3-methylimidazolium hexafluorophosphate (BmimPF₆) as the P-precursor.²³ Jia *et al.* have synthesized P-doped C₃N₄ using phosphoric acid as the P-source, which shows a H₂ activity rate of 256.4 μmol h⁻¹, which is almost 24 times higher than that of pristine C₃N₄.²⁴ Yuan *et al.* have reported that P-doped C₃N₄, synthesized using melamine and (hydroxyethylidene)diphosphonic acid (HEDP) precursors, showed a H₂ evolution activity of 104.1 μmol h⁻¹, which is 9.2 times higher than that of undoped C₃N₄.²⁵ Other

phosphorous precursors reported to synthesize P-doped C₃N₄ are ammonium hexafluorophosphate, diammoniumhydrogen phosphate, phosphoric acid, diphosphonic acid, red P *etc.*²⁶ It is to be noted that the majority of the earlier studies on P-doped C₃N₄ are related to photocatalytic H₂ evolution in the presence of SEDs such as TEOA and methanol, thus under-utilizing the efficacy of the photogenerated holes. Thus, the improved lifetime of photocarriers upon doping was not harvested effectively as the studies focus only on reductive half-reactions. Additionally, unlike most of the earlier reports, the present study focuses on the use of eco-friendly inositol hexaphosphate (IP6), commonly known as phytic acid as a P precursor. Owing to its high abundance, biocompatibility and most importantly the environmental-friendly aspects, IP6 is believed to be a suitable candidate in comparison to majority of the P-precursors reported in the literature to prepare P-doped C₃N₄. Due to the presence of six phosphate groups and twelve ionizable protons of various acidic strengths, IP6 is capable of interacting with C₃N₄-precursors such as dicyandiamide (DCDA) quite effectively. There exist few reports on the use of phytic acid as the P-precursor for the synthesis of P-doped C₃N₄; however, the photocatalytic studies are limited in exploring reductive half reactions, *i.e.*, photocatalytic H₂ evolution activity, in addition to the use of multiple time-consuming steps to yield P-doped C₃N₄.^{13,27} The development of dual-functional semiconductors for effective utilization has recently attracted attention.⁸ Yu *et al.* have demonstrated the simultaneous production of H₂ and BADH from BA oxidation using CN/BP@Ni heterostructure nanosheets with the activity of H₂ and BADH as 0.928 μmol h⁻¹ and 0.93 μmol h⁻¹ respectively.²⁸ Zhang *et al.* reported dual-functional, S or P-doped C₃N₄ using tri-thiocyanuric acid or melamine polyphosphate, respectively.²⁹ Chen *et al.* have synthesized P-doped Zn_xCd_{1-x}S for hydroxyl methylfurfural (HMF) oxidation and H₂ production simultaneously with the activity of H₂ production increased to 786 μmol h⁻¹ g⁻¹ with 2,5-diformylfuran (DFF) as a value-added oxidation product.³⁰

In the present study, we report the doping of P-atom in the g-C₃N₄ matrix using environmentally friendly inositol hexaphosphate (IP6), commonly known as phytic acid. The enhanced visible-light photocatalytic activity and efficient harvesting of hole–electron pairs of P-doped C₃N₄ were studied for the simultaneous production of H₂ and BADH.

Results and discussion

P-doped graphitic carbon nitride (CN-*x*P) samples with various amounts (*x*) of phosphorous (P) dopants were prepared *via* the thermal condensation of dicyandiamide (DCDA) and inositol hexaphosphate (IP6) that acted as the source of P. Various amounts of P-dopants were introduced into the C₃N₄ lattice by tuning the amount of IP6 in the precursor mixture. It is noted that an intermediate complex is formed when DCDA reacts with IP6, due to electrostatic interactions. The obtained powders after solvent evaporation were heated at 550 °C to obtain CN-*x*P samples (Fig. 1A). The as-obtained powders of various CN samples (doped and undoped) were analysed by powder X-ray diffraction (PXRD) technique to investigate the crystal



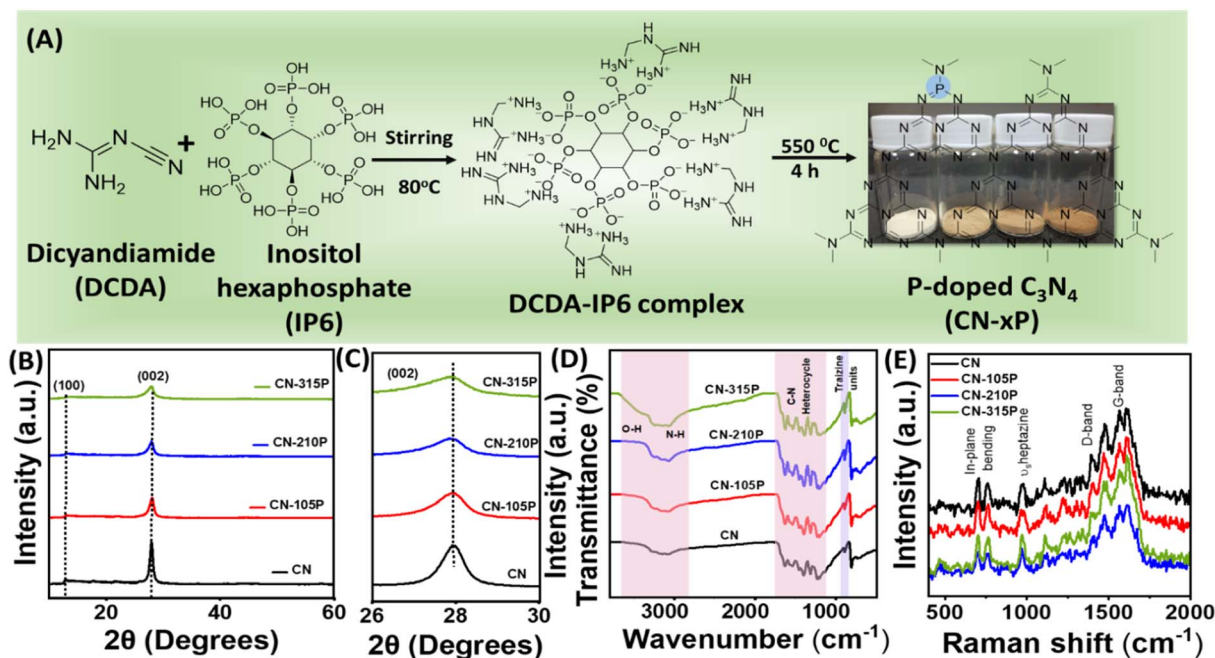


Fig. 1 (A) Schematic representation of the preparation of P-doped carbon nitride, (B and C) XRD patterns, (D) FTIR and (E) Raman spectra of pristine and P-doped carbon nitride. Digital pictures of CN-xP powders are also shown in (A).

structure. As shown in Fig. 1B, the PXRD pattern of CN shows two distinct diffraction peaks at 2θ values of 27.9° and 12.9° , which are indexed to the (002) and (100) planes, respectively (JCPDF no. 87-1526).^{15,16} The intense (002) diffraction peak observed at around 27.9° represents the inter-planar graphitic stacking and the diffraction peak observed at 12.9° corresponds to the (100) plane due to the in-plane tri-*s*-triazine packing motif.¹³ A noticeable shift in the diffraction peak at 27.9° towards the lower 2θ values is observed for CN-xP samples in comparison to undoped CN, indicating an increase in interlayer distance by lattice expansion along the *c*-axis direction due to the presence of bulkier P dopants (100 pm), which replace relatively smaller C atoms (70 pm) (Fig. 1C).¹⁶ In addition, the broadening of diffraction peaks with decreased intensity observed in CN-xP suggests a disordered structure due to the incorporation of bulkier P in the C_3N_4 matrix. The peak at 12.9° shifts to a lower 2θ value, which indicates an increase in planar nitride pores.¹³ The FTIR spectra of CN and CN-xP are shown in Fig. 1D. The band located at 803 cm^{-1} is assigned to the out-of-plane breathing vibrations of tri-*s*-triazine units. The observed bands ranging from 3000 cm^{-1} to 3500 cm^{-1} appear due to water molecules adsorbed on samples and uncondensed amine groups. The bands observed between 1200 cm^{-1} and 1600 cm^{-1} are assigned to the stretching vibrational modes of CN heterocycles.¹⁷ Further, the band located at around 801 cm^{-1} for CN shifts to a higher wavenumber in the case of CN-xP samples. Moreover, the decrease in the intensity of the bands observed in the range of 3000 cm^{-1} to 3300 cm^{-1} as a function of P-amount suggests successful incorporation of P into the CN matrix,¹⁷ which is in agreement with the XRD data. Peaks related to P-containing species are not observed, probably due to the low

phosphorus contents present in the C_3N_4 lattice.¹⁷ The formation of carbon nitride was further confirmed by Raman spectroscopy. As shown in Fig. 1E, all characteristic Raman bands corresponding to C_3N_4 are intact in CN-xP, suggesting that the basic structure of C_3N_4 is unaltered even after P doping. The bands observed at 980 cm^{-1} and 690 cm^{-1} originate due to the symmetric N-breathing mode and in-plane bending modes of heptazine units respectively. The bands located between 1200 and 1700 cm^{-1} are ascribed to the disordered graphitic carbon nitrogen vibrations.¹⁸ The band located at around 1570 cm^{-1} is due to the G band of C_3N_4 , which is related to the degree of graphitization while the D band located at around 1398 cm^{-1} is related to the structural defects and partially distorted structures of C-sp². The ratio of the intensity of D-band (I_D) to that of G-band (I_G) is known to provide information about the structural distortion of CN. It is observed that the value of I_D/I_G is high for P-doped samples as compared to the undoped sample, suggesting a relatively more structural disorder in case of P-doped samples (Table S1†).³¹ The morphology of CN and CN-210P samples was characterized by field emission scanning electron microscopy (FESEM) and transmission electron microscopy (TEM) techniques. The FESEM images of CN and CN-210P indicate the presence of layered morphology (Fig. 2A and B). As evident from the images, CN-210P consists of loosely bound thin layers as compared to that of CN, where it consists of bulk aggregated particles. The FESEM images of CN-105P and CN0315P are given in Fig. S1.† The TEM images further confirm layered morphology composed of thin nanosheets (Fig. 2C and D). The high-resolution TEM (HRTEM) of CN-210P shows well-resolved lattice fringes that correspond to the (002) plane of C_3N_4 as the spacing between the fringes is found to be



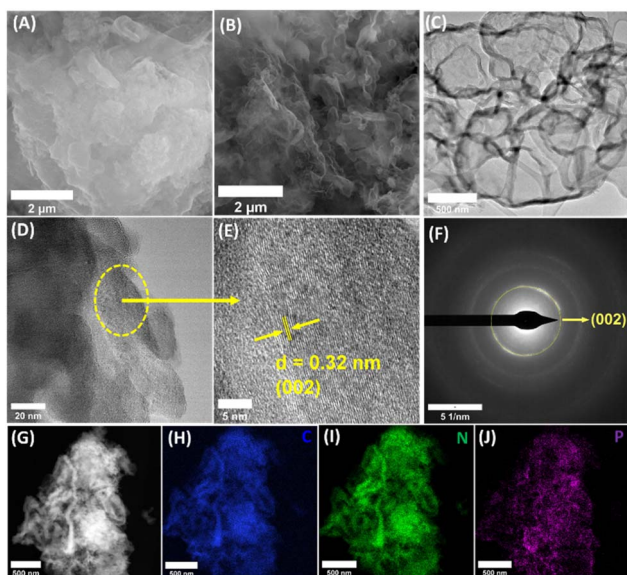


Fig. 2 FESEM images of (A) CN and (B) CN-210P. (C and D) TEM images of CN-210P at different magnifications. (E) HRTEM, (F) SAED, (G) STEM-HAADF images of CN-210P. (H–J) STEM elemental mapping corresponding to C, N and P, scale bar 500 nm.

≈ 0.32 nm (Fig. 2E). Further, the ring pattern observed in the SAED pattern suggests the poly crystallinity of CN-210P and the diffraction ring can be indexed to the (002) reflection (Fig. 2F), which is consistent with HRTEM and XRD data. The presence and homogeneous distribution of P-dopants across CN-210P nanosheets are clearly evident from the elemental mapping obtained using STEM-EDS (Fig. 2G–J).

The survey spectra of the samples reveal the presence of corresponding elements (Fig. S2†). The C 1s core level XPS data of CN-210P can be deconvoluted into three peaks (Fig. 3A) namely, C1 observed at a binding energy (BE) of 288.0 eV, C2 at 286.2 eV and C3 at 284.8 eV, which are attributed to the carbon atom of the heptazine ($\text{N}=\text{C}(\text{N})_2$) group, C–NH_x and graphitic carbon (C–C) respectively.²⁷ The N 1s XPS spectra of the CN and CN-210P samples (Fig. 3B) can be fitted into four distinct peaks, which are attributed to C–N=C (N1, 398.6 eV), N–C₃ (N2, 400.5 eV), C–NH₂ (N3 401.5 eV amino functional group) and for charging effects or π -excitations in heterocycles (N4, 404.7 eV).²⁷ It is noteworthy that the peak positions observed for CN-210P are slightly shifted to higher BE values than those of CN, indicating that P-doping modifies the electronic structure C₃N₄.²⁷ The P 2p core level spectra of the CN-210P sample show a characteristic peak at 133.6 eV, which corresponds to the P–N linkage^{13,27} (Fig. 3C). Additionally, the P 2p XPS data of CN-210P do not exhibit any peak corresponding to the P–C bonding, which usually appears at 1–2 eV lower than P–N bonding, suggesting that the P-dopant most probably replaces the C atom in the C–N framework of g-C₃N₄, thus forming P–N bonds.¹³ The O 1s spectrum of CN-210P shows two peaks located at BE values of 532.6 eV and 533.8 eV, which corresponds to adsorbed water and O–P linkages, respectively (Fig. S3†).³² The undoped and P-doped samples were analysed using solid-state nuclear

magnetic resonance (SSNMR) measurements. As shown in Fig. S4A,† the ¹³C NMR spectra of CN and CN-210P show two prominent peaks at 153 ppm and 162.3 ppm, originating due to the carbon atoms of N–C=N (C_{3N}) and –C–NH_x groups (C_{2N-NH_x}), respectively.³³ The absence of additional peaks confirm that carbon atoms are not directly linked to P-dopants. In other words, ¹³C NMR spectra reveal the non-existence of P–C linkages in P-doped samples, supplementing XPS results. The ratio of intensity of peaks due to C_{3N} and C_{2N-NH_x} ($I_{\text{C}_{3\text{N}}}/I_{\text{C}_{2\text{N-NH}_x}}$) may provide insights into the doping of P in heptazine rings. It is noted that the value of $I_{\text{C}_{3\text{N}}}/I_{\text{C}_{2\text{N-NH}_x}}$ is found to be 0.37 for CN, while it is 0.43 for CN-210P, suggesting that P-doping is attained by preferential replacement of the C-site of C_{2N-NH_x}. Further, the shifts in the peak positions in the case of CN-210P as compared to CN may be due to the presence of P-dopants, which may change the electron density and local chemical environment.³⁴ The ¹⁵N NMR spectrum (Fig. S4B†) of CN-210P and CN shows four signals at 191.3, 152.3, 132.5 and 112.4 ppm corresponding to C=N–C(N1), N3C (N2), C–NH–C (N3) and NH₂ (N4) respectively.³³ The incorporation of P-dopants into g-C₃N₄ can be clearly seen in the ³¹P NMR spectrum that show four signals between 0 and -50 ppm (Fig. 3E), attributed to four possible different positions in the framework structure. Thus, it is envisaged that P-dopants can have four possibilities to replace either bay or corner C atoms.³⁵ Among bay and corner carbon atoms, the P-atom prefers to replace the corner C atom rather than the bay C atom, as replacing the former is reported to be relatively easy than the latter.^{13,35} The substitution of corner C atoms in the tri-s-triazine unit by the P atom is known to result in the elongation of in-planar tri-s-triazine motifs, complementing XRD data, as shown in Fig. 1B. In order to validate this, DFT calculations were performed. The single-layer g-C₃N₄ structure was optimized and its cell parameter was found to be 6.93 Å. A 2 × 2 × 1 super cell consisting of 32 N and 24 C atoms was considered to model the P doped system with a vacuum layer of 15 Å, and the corresponding optimized geometry is reported in Fig. S5.† Substitutional doping of P at different N (N1, N2 and N3) and C (C1 and C2) sites, as depicted in Fig. 3D and S5,† was considered and their optimized structures along with the formation energies are reported in Fig. 3D and S5.† In the case of P doped at the N-site of g-C₃N₄, the doping at the N1 site which is connecting three heptazine units is found to be the minimum energy configuration while for the doping at the C-site, the corner C (C1)-site is the minimum energy one, which is consistent with the earlier reports.^{13,14} From the minimum energy structures, P-doping energy at C1 and N1 sites were calculated to be -0.27 eV and 1.70 eV respectively, clearly indicating that the P-doping at C1-site is exothermic while it is endothermic at all the three N-sites, in consistence with the experimental observations. In the optimized structure of P-dopants at the C1-site, the three P–N bond distances were measured to be 1.613 Å, 1.624 Å and 1.722 Å. The CHNS elemental analysis data of doped and undoped samples indicate that the molar ratio of C/N for all samples was found to be around 0.66 (Table S2†), similar to the values reported earlier.³³ The surface area and pore size distribution of the samples were measured using the Brunauer–Emmett–Teller (BET)



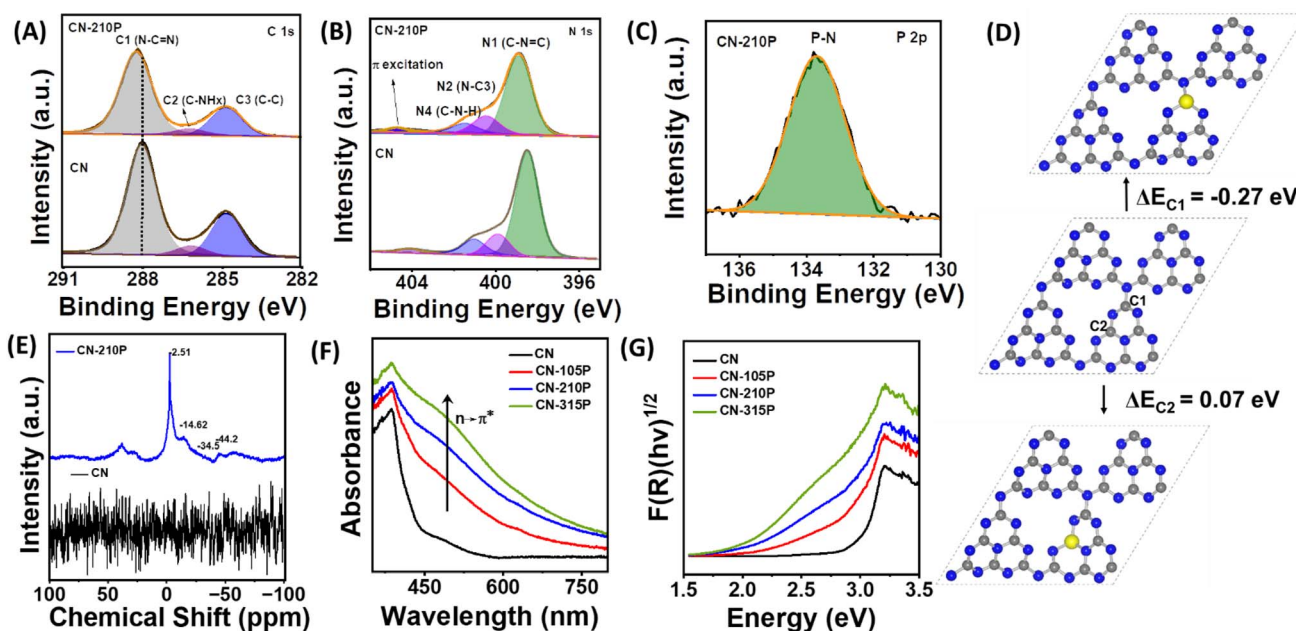


Fig. 3 XPS core-level spectra corresponding to (A) C 1s, (B) N 1s, and (C) P 2p of pristine and CN-210P. (D) Optimized structures of the $2 \times 2 \times 1$ super cell of pristine C_3N_4 and P-doped systems at different possible C-sites with the corresponding formation energies. (E) ^{31}P NMR spectra of CN and CN-210P. (F) UV-vis DRS and (G) Tauc plot of CN and CN-xP.

measurements. As evident from Fig. S6,[†] all the prepared materials exhibit type-IV adsorption.³³ The specific surface areas of CN-xP samples are slightly lower than those of the pristine CN. It has been reported that doping usually modifies the electronic structure and the structural distortion caused by doping may result in a reduction in the surface area, as reported earlier.³⁶ The pore size of CN-210P (46.2 nm) was found to be higher than that of CN (43.3 nm). It is anticipated that larger pores of the mesoporous structure are beneficial for facile diffusion of reactants and efficient light capture.³⁷ Parameters such as specific surface area, average pore volume and total pore volume of all the catalysts obtained from BET data are tabulated in Table S3.[†] Further, the effect of doping on the absorption characteristics of C_3N_4 was studied by UV-vis diffuse reflectance spectroscopy (UV-vis DRS) measurements. The samples showed apparent visual color change from light yellow to brown upon increasing the P content in the P-doped samples (Fig. 1A). As displayed in Fig. 3F, extended visible light absorption is noted in the case of P-doped samples as compared to that of the undoped sample, indicating enhanced light-harvesting ability of CN-xP samples. A gradual shift in the absorption edge to higher wavelengths is noted with gradual increase in the P content present in C_3N_4 , suggesting bandgap engineering upon P-doping. The optical bandgap (E_g) of all the samples was measured from the Tauc plot, as shown in Fig. 3G and S7.[†] The values of E_g for CN, CN-105P, CN-210P, and CN-315P were found to be 2.98 eV, 2.82 eV, 2.61 eV and 2.45 eV, respectively. The observed E_g values portray the narrowing of E_g due to the P-doping of C_3N_4 . Moreover, a faint shoulder around 500 nm observed in the absorption spectra of doped samples is known to originate due to the relatively disordered C_3N_4

framework wherein $n \rightarrow \pi^*$ excitation of lone pair electrons of edge N becomes allowed unlike the case of planar C_3N_4 where it is forbidden. This assists in the absorption of photon energy in the visible region.³⁸ Additionally, the P-dopant results in the creation of sub-band gap states (E_T), as indicated in Fig. S7 and S8.[†] The P-dopant-induced changes in the band structure of C_3N_4 were analysed by locating the positions of the VB and CB of P-doped and undoped CN samples using UV-vis DRS data in combination with either XPS VB spectra or Mott-Schottky (M-S) measurements. Using eqn (1),³³ the position of the VB *vs.* normal hydrogen electrode (NHE) was estimated from the valence band maxima (VBM), which was determined from the XPS-VB spectra (Fig. 4A).

$$E_{VB-NHE} = \phi + E_{VB-XPS} - 4.44 \quad (1)$$

where E_{VB-NHE} is the VB potential *vs.* NHE, ϕ is the electron work function of the XPS analyser and the value is 4.21 eV in the present case, and E_{VB-XPS} is the VBM value obtained from the XPS-VB spectra. From the as-obtained VB values and E_g values determined from UV-vis DRS data (Fig. 3G), the positions of the CB of CN and CN-xP were estimated ($E_{CB} = E_{VB} - E_g$) and are schematically shown in Fig. S8.[†] Similarly, the VB and CB positions determined using Mott-Schottky measurements (Fig. S9[†]) are comparable to the values obtained from the VB-XPS spectra (see ESI for more details, Tables S4 and S5[†]).

Impressed by the enhanced light absorption and suitable VB and CB positions of CN-xP, photocatalytic activity studies were performed using a custom-built white LED light source. The spectral output of the LED used in the present study is given in Fig. S10.[†] Prior to dual functional photocatalytic activity studies, H_2 evolution studies for all the photocatalysts (doped



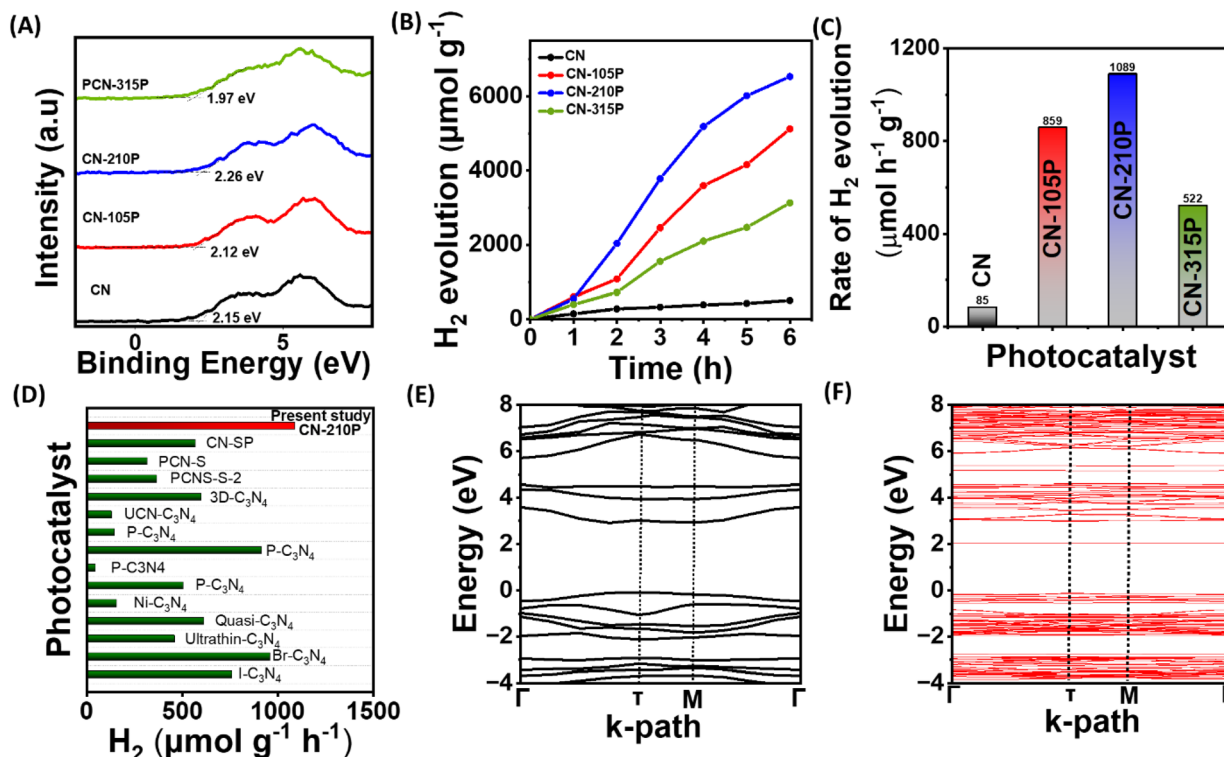


Fig. 4 (A) XPS VB spectra. (B) Photocatalytic hydrogen evolution as a function of time using TEOA as the SED. (C) Hydrogen evolution rate obtained using the CN and CN-xP sample. (D) Comparison plot depicting the H₂ evolution activity of CN-210P with some of the reported systems. Electronic band structure of (E) g-C₃N₄ and (F) P-doped C₃N₄ from the HSE06 functional.

and undoped) with photodeposited Pt co-catalysts were carried out in the presence of triethanolamine (TEOA) as the SED. It is to be noted that the reflections corresponding to Pt are not seen in PXRD data, indicating high dispersibility of small Pt particles on the CN-210P surface. The TEM data also support high dispersibility of Pt particles on CN-210P (Fig. S11†). Fig. 4B depicts the photocatalytic H₂ evolution performance of undoped and P-doped CN consisting of different amounts of P. As observed from Fig. 4C, around 6 to 12.8 times increase in the H₂ production rate was observed in the case of P-doped samples depending on the P content, as compared to undoped C₃N₄. For instance, CN-210P exhibits a H₂ production rate of around 1089 μmol h⁻¹ g⁻¹, 12.8 times higher than that of undoped CN (85 μmol h⁻¹ g⁻¹), suggesting better charge separation ability and availability of more reactive sites in P-doped samples unlike pristine CN. It is noteworthy that the activity is tunable with the P-content present in the C₃N₄ lattice. The photocatalytic activity increases with the increase in the P-content; however, beyond a certain threshold (Fig. 4B and C), the activity decreases due to the presence of excess P dopants, which may themselves act as recombination centres and thus inhibit the charge carrier separation.²⁷ Among all the P-doped catalysts, CN-210P with a P-dopant quantity of 0.66 at% (from XPS) appears to be the optimum quantity and shows the highest rate of H₂ production among other studied catalysts. The enhanced visible light absorption and associated narrow band gap are responsible for the observed higher activity of CN-xP than that of CN. Table S6†

and Fig. 4D show the comparison of the photocatalytic activity of CN-210P with other reported materials. These results indicate the appreciative effect of P that induces electronic structure modifications, which assist in increasing photocatalytic activity. The P atoms that replace C-sites in g-C₃N₄ form P-N bonding with the adjacent N atoms, leading to disorder in the heptazine skeleton. It has been reported that these P-N bonds in P-doped C₃N₄ are known to break the localized state around the highly symmetric heptazine units of undoped C₃N₄, leading to the creation of charge carrier-rich regions that facilitates charge transfer. The extra non-bonding electron of P dopants will be delocalized in the tri-s-triazine ring, creating P⁺ centres, and thus enhanced electron-rich tri-s-triazine moieties present in P-doped C₃N₄.³⁹ Notably, the HOMO and LUMO localized around P atoms and neighbouring N atoms, which facilitates the movement of photogenerated carriers under visible light between neighbouring heptazine units through the N-P-N-C channel.¹⁴

Further, DFT calculations were performed to understand the electronic structure and its manifestation on P-doping in g-C₃N₄. The electronic band structure and density of states (DOS) of both pure and P-doped C₃N₄ were calculated using PBE and HSE06 functionals. The band structure plots shown in Fig. S12† indicate that the electronic band gap of C₃N₄ has been significantly underestimated to 1.86 eV from the PBE functional, while the corresponding band gap determined from HSE06 is found to be 3.01 eV, which is close to the experimental value (Fig. 4E).



From the projected DOS (PDOS) reported in Fig. S13,[†] it is observed that the valence band maxima (VBM) is mostly contributed by the N 2p states while the conduction band minima (CBM) is coming from C 2p and N 2p states, which can also be seen from the band decomposed charge density iso-surface plots reported in Fig. S14.[†] The band structure of P-doped C₃N₄ shows an isolated mid-gap state around 1 eV below the CB resulting in a reduction in the band gap from 3.01 eV in the pure system to 2.13 eV (Fig. 4F), which is consistent with the observed improved photoactivity after P-doping. From the PDOS reported in Fig. S13B[†] and the band decomposed charge density iso-surface shown in Fig. S15C,[†] it can be observed that the mid-gap state arises from the P 2p, N 2p and C 2p states. From the charge density iso-surface plots, it can also be observed that both the highest single electron occupied band (Fig. S15B[†]) and the mid-gap band (Fig. S15C[†]) are majorly contributed by the P and the N atoms bound to P. It is also clear from these band decomposed charged densities that in the case of pure systems, the VBM and CBM are almost uniformly distributed over the cell while in the P-doped system, better spatial separation of charge carries can be expected.

The apparent quantum efficiency (AQE) of the optimized catalyst was determined (See ESI). The AQE was found to be 1.6%, 1.01% and 0.16% at 452, 517 and 636 nm respectively. As shown in Fig. S16,[†] the wavelength-dependent AQE values determined for CN-210P are consistent with the absorption

spectrum of CN-210P. The AQE values obtained in the present study is comparable to many of the reported photocatalysts (Fig. S16, Table S7[†]).

The suitability of CN-xP as a dual-functional catalyst was demonstrated by carrying out photocatalytic studies that produce H₂ and benzaldehyde, simultaneously. In other words, we utilized photogenerated electrons to produce H₂ and photogenerated holes to produce BADH, an approach wherein the charge carriers are harvested effectively without wasting the oxidizing ability of holes (Fig. 5A). The H₂ production from photocatalytic studies conducted in the presence of BA obtained using CN-xP is shown in Fig. 5B. The P-doped C₃N₄ (CN-210P) shows 3 times higher rate of H₂ production than that of pristine CN, confirming the determining role of P in dictating the photocatalytic activity. Among all the CN-xP samples, CN-210P exhibited better activity (Fig. 5B and C). On prolonged illumination for 12 h, CN-210P yields appreciable amounts of H₂ (Fig. 5D). Further, the products formed during the photocatalytic oxidation of BA were monitored using various techniques such as ¹H NMR, liquid chromatography (LC), and gas chromatography-mass spectrometry (GC-MS). Fig. S17[†] shows the ¹H NMR spectra recorded for the reaction mixture obtained using CN and CN-xP samples collected after continuous illumination of light for 7 h. As shown in the figure, the peaks related to BADH are clearly evident, confirming the selective oxidation of BA. It should be noted that the yield of BADH using

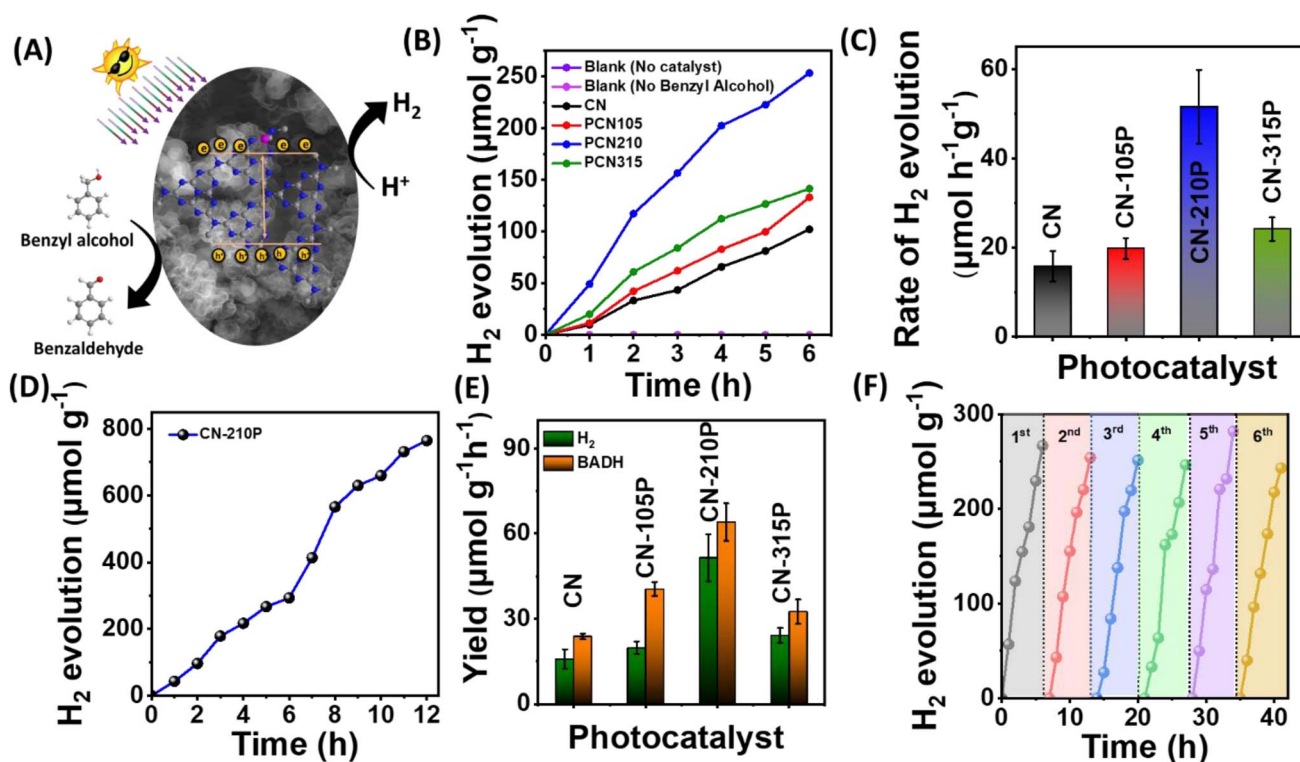


Fig. 5 (A) Schematic depicting the dual functionality of CN-xP for the simultaneous production of H₂ and benzaldehyde. (B) Hydrogen evolution studies in the presence of 20 mM benzylalcohol. (C) Rate of hydrogen evolution using CN and CN-xP. (D) Hydrogen evolution observed using CN-210P as the photocatalyst for longer durations. (E) Yields of H₂ and benzaldehyde obtained using CN and CN-xP. (F) Cyclic runs of H₂ evolution over CN-210P.



CN-210P was found to be $57.3 \mu\text{mol h}^{-1} \text{g}^{-1}$, which is 2.8 times higher than that of CN (Fig. 5E). Similarly, LC and GC-MS data also revealed the formation of BADH as the only oxidation product formed during the photocatalytic reaction, indicating good selectivity (Fig. S18 and S19†). Overall, the CN-210P sample demonstrates excellent dual-functional photocatalytic activity for the simultaneous production of H_2 and BADH. It is to be noted that CN-210P is capable of producing H_2 and BADH even in the presence of natural sunlight, indicating the applicability of this material in practical applications (Fig. S20†). Further, the positions of VB and CB are appropriately located with respect to proton reduction and BA oxidation to realize proton reduction by photogenerated electrons and BA oxidation by photogenerated holes.⁴⁰ Additionally, the sub-band states originated due to P-doping located near the Fermi level also contributes to the enhanced activity in case of P-doped C_3N_4 .⁴¹ To gain further insights, a range of control experiments were conducted. The experiments conducted in the absence of BA do not result in the production of H_2 and BADH. Similar observation has been noted when CN-210P was used as a photocatalyst under dark conditions, co-catalyst free conditions, as well as photocatalyst-free conditions, suggesting the essential role of light and photocatalysts. Notably, experiments were conducted to probe the reactive species that participated in photocatalysis, by adding tetrachloromethane (CCl_4) and TEOA to the reaction mixture, as these are known to act as scavengers for electrons (e^-) and holes (h^+) respectively. A remarkable decrease in H_2 evolution rate was noted, when CCl_4 or $\text{K}_2\text{S}_2\text{O}_8$ was used in the reaction mixture, indicating that the proton reduction process was driven by photogenerated electrons (Fig. S21A†). With the addition of AO or TEOA, the yield of BADH decreased remarkably, suggesting that photogenerated holes drive oxidation of BA to BADH (Fig. S21B†). Thus, it can be concluded that CN-xP harvests electron-hole pairs effectively in concomitant production of H_2 and BADH. Further, when the measurements were carried out in the presence of isopropyl alcohol (IPA), which is known to act as a scavenger for $\cdot\text{OH}$ radicals, the yields of H_2 and BADH remain unchanged, suggesting that $\cdot\text{OH}$ radicals may not be involved in the conversion of BA to BADH (Fig. S21A and B†). Notably, when experiments were carried out in a pure acetonitrile medium (water-free conditions), no H_2 was observed (minor trace), indicating that water plays a key role in H_2 production. In addition, increased amounts of H_2 evolution are noted when the concentration of BA is increased. These measurements indicate that water and BA mutually promote the photocatalytic H_2 production. Additionally, electron spin resonance (ESR) measurements were carried out to further understand the conversion of BA into BADH (Fig. S22†). As shown in Fig. S22,† the ESR spectrum does not show any signals when the reaction is carried out in the darkness, whereas evident ESR signals are noticed when the reaction is carried out in the presence of light. Based on these observations, it is envisaged that the conversion of BA to BADH proceeds *via* a carbon-centred radical route, as proposed in Fig. S23.† The DFT calculations also support the experimental observations (see ESI†).

The robustness of CN-210P was examined by performing photocatalytic measurements for repeated cycles and long hours of continuous illumination. As shown in Fig. 5D, CN-210P exhibited excellent stability when the measurements continued for 12 h of illumination. Notably, the activity of CN-210P towards H_2 generation remains unaltered for six cycles, that is, for 36 h (Fig. 5F). In order to confirm the robustness of CN-210P, the catalyst after the photocatalytic studies was collected and characterized by FTIR spectroscopy, XRD, XPS and Raman spectroscopy measurements. As shown in Fig. S26–S28,† insignificant changes are noted even after repeated use of the photocatalyst in photocatalytic reactions, suggesting the stability of CN-210P.

To understand the observed enhanced photocatalytic activity of P-doped samples, steady-state photoluminescence (PL) and time-resolved PL (TRPL) measurements were carried out. As shown in Fig. 6A, the PL spectra of undoped CN recorded at an excitation wavelength of 325 nm exhibit a broad peak at around 490 nm, which shifts to a higher wavelength in the case of P-doped CN, suggesting the narrowing of bandgap upon P-doping.⁴² Additionally, the data indicate that the PL intensity is quenched in P-doped samples as compared to the pristine CN, suggesting suppressed recombination of the photo-induced charge carriers in P-doped samples. Besides, the lifetime of photogenerated charge carriers was determined from the TRPL spectra, as shown in Fig. 6B. The PL decay curves recorded at 296 nm of doped and undoped CN samples were fitted using the biexponential decay function (eqn (2)):

$$I(t) = A_1 e^{-\frac{t}{\tau_1}} + A_2 e^{-\frac{t}{\tau_2}} + B \quad (2)$$

where A_1 and A_2 are the relative amplitudes, $I(t)$ is the intensity of TRPL decay, and τ_1 and τ_2 are the short and long fluorescence lifetimes respectively.

The relative% of the pre-exponential factor was calculated using eqn (3) and (4):

$$A_1\% \text{ relative} = \frac{A_1 \tau_1}{A_1 \tau_1 + A_2 \tau_2} \quad (3)$$

$$A_2\% \text{ relative} = \frac{A_2 \tau_2}{A_1 \tau_1 + A_2 \tau_2} \quad (4)$$

The average PL lifetime (τ) was calculated using eqn (5):

$$\tau = \frac{A_1 \tau_1^2}{A_1 \tau_1} + \frac{A_2 \tau_2^2}{A_1 \tau_2} \quad (5)$$

The average decay lifetime of P-doped samples (CN-xP) is found to be smaller than that of pristine CN, reaffirming the enhanced charge separation efficiency and rapid charge carrier migration in CN-xP than CN. It is anticipated that the additional migration channels arising due to P–N linkages in CN-xP samples enable faster charge transfer.^{15,27} Table S8† summarizes the average decay life time of CN-xP and CN samples determined from the TRPL data. Further, CN and CN-210P powder samples were characterized using ESR recorded at



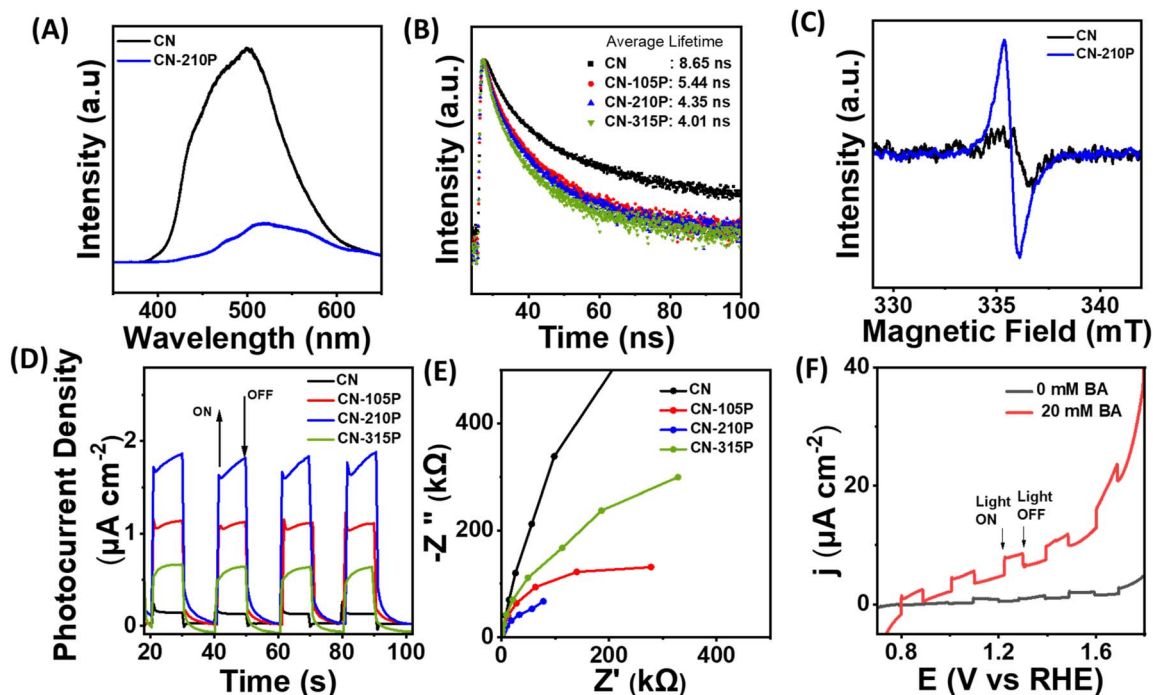


Fig. 6 (A) Steady-state photoluminescence of CN and CN-210P. (B) Time-resolved photoluminescence of CN-xP. (C) ESR spectra of CN and CN-210P solid powders, recorded at room temperature. (D) Photoelectrochemical measurements depicting photocurrent recorded at 1.23 V vs. RHE. (E) EIS data obtained at 1.23 V vs. RHE. (F) LSVs recorded in the presence and absence of BA.

room temperature. As shown in Fig. 6C, the samples exhibited a single Lorentzian line, originating due to the presence of unpaired electrons of π -conjugated aromatic rings of $g\text{-C}_3\text{N}_4$. Notably, the intensity of the ESR signal of CN-210P is enhanced as compared to that of CN, suggesting high concentration of delocalized electrons on π -conjugated aromatic rings of CN-210P. Thus, the P-doping in C_3N_4 leads to the formation of electron-rich C_3N_4 . The delocalized electrons on π -conjugated aromatic rings are known to facilitate the charge transfer, and enhanced photocatalytic activity can be expected in P-doped samples. To further validate the improved life time of photo-generated charge carriers, photoelectrochemical (PEC) measurements were carried out. The photocurrent responses were recorded at 1.23 V vs. RHE by chopping the light at regular intervals of time. As shown in Fig. 6D, around 1.6 times increase in photocurrent density is noted with CN-210P in comparison to CN, indicating the relatively improved electron-hole separation. It is expected that the dopants (P) can act as trap centres and minimize the recombination of photogenerated charge carriers.²⁷ The electrochemical impedance spectroscopy (EIS) studies indicate smaller charge transfer resistance (R_{ct}) of P-doped samples, suggesting their good charge transfer ability at the electrode/electrolyte interface (Fig. 6E). Further, as shown in Fig. 6F, the higher photocurrents observed from the voltammograms recorded in the presence of BA clearly indicate the oxidative ability of CN-210P towards BA.^{43a} Additionally, the electrochemical hydrogen evolution reaction (HER) polarization curves recorded in the darkness suggest that the overpotential to reach 0.1 mA cm^{-2} is 0.47 V vs. RHE for CN-210P, while it is

0.54 V vs. RHE for CN (Fig. S29). This again confirms the beneficial role of the presence of P-dopants in achieving high HER activity.

Conclusions

In summary, P-doped carbon nitride with different amounts of P dopants has been prepared using a mixture of DCDA and eco-friendly inositol hexaphosphate (IP6) as the P-source. Enhanced light harvesting due to the narrowing of the band gap of C_3N_4 upon P-doping was demonstrated. The systematic variations in the band structure of C_3N_4 with dopant (P) quantity were studied. DFT studies revealed that the doping of P into C_3N_4 occurs preferentially through the replacement of the corner carbon atom. Among all the P-doped CN samples, CN-210P exhibited high photocatalytic activity for H_2 evolution as compared to undoped CN. Around 12.8 times higher H_2 production rate was observed in the case of CN-210P than that of pristine CN when TEOA was the SED. The appreciable enhancement in photocatalytic activity in CN-210P is due to the result of P-doping, which induced modifications in the electronic structure that enable wide visible light absorption and provide more active sites, as suggested by DFT calculations. The presence of P-N linkages acts as additional migration channels and facilitates efficient charge transfer. Most importantly, it is confirmed that CN-210P can act as a dual functional catalyst for producing H_2 and BADH simultaneously. The yield of H_2 and BADH was found to be $51.6 \mu\text{mol h}^{-1} \text{ g}^{-1}$ and $64.2 \mu\text{mol h}^{-1} \text{ g}^{-1}$ respectively. ESR measurements and DFT calculations



suggested that the photocatalytic oxidation proceeds *via* a carbon-centred radical route to selectively yield BADH. The study provides opportunities to explore the ability of CN-*x*P for oxidizing other biomass-derived platform molecules as SEDs to simultaneously produce H₂ and high-value products.

Experimental

Chemicals

All chemical reagents were purchased from commercial suppliers and used directly as received without further purification. The chemicals used are as follows: dicyandiamide (DCDA) (Loba Chemie, 98%), inositol hexaphosphate (50% in water, TCI Chemicals), benzyl alcohol (Loba Chemie, 99%), benzaldehyde (Sigma-Aldrich, > 99%), triethanolamine (Loba Chemie), methanol (Molychem 99%), H₂PtCl₆ · 6H₂O (Sigma-Aldrich, 37.50% Pt basis), sodium sulfate (Molychem, 99%), and maleic Acid (Loba Chemie, 99%).

Preparation of P-doped carbon nitride (CN-*x*P)

A precursor mixture containing a known amount (5 g) of DCDA and a certain amount of inositol hexaphosphate (IP6) was added to 4.55 mL water, followed by evaporation of the solvent by heating at 80 °C. Subsequently, the white solid obtained was transferred into a cylindrical crucible of volume 30 mL covered with a lid, kept in a muffle furnace, and heated at 550° for 4 h at a ramping rate of 5 °C min⁻¹. After natural cooling to room temperature, the products were obtained and hand-milled before use. CN with different amounts of P-dopants (CN-*x*P) was obtained by varying the amounts of IP6 in the precursor mixture, where *x* represents the amount (in μL) of IP6 taken in the precursor mixture.

Preparation of bulk g-C₃N₄ (CN)

A conventional calcination method was used to prepare pristine g-C₃N₄ (CN). A known amount (5 g) of DCDA was taken in a 30 mL cylindrical crucible covered with a lid, transferred into a muffle furnace and heated at 550° for 4 hours at a ramping rate of 5 °C min⁻¹. After cooling to room temperature naturally, the product was milled into a fine powder, washed with water and dried overnight to obtain the final CN.

Characterization

The powder XRD patterns were recorded using a Bruker D8 Advance with Cu K α radiation ($\lambda = 1.5406 \text{ \AA}$). The FESEM images were recorded using a Quanta FEG 250. The samples for SEM were prepared by drop-casting the dilute dispersions of samples onto a pre-cleaned Si substrate. The Fourier transform infrared (FTIR) spectra were recorded using a PerkinElmer spectrophotometer (Model: spectrum two) in the ATR mode. The surface area measurements were carried out using a quanta chrome NOVA 1000e BET surface area analyser. The X-ray photoelectron spectroscopy (XPS) characterization was performed using a K-alpha Thermo Fisher Scientific Instrument with an Al K α X-ray source (1486.6 eV). The UV-visible diffuse reflectance spectra (UV-vis DRS) were recorded using a UV-

visible Jasco V770 spectrometer with an integrated sphere setup and BaSO₄ as a reference. Raman and PL spectra were recorded using a LABRAM HR, Horiba France at an excitation wavelength of 325 nm. The photoelectrochemical measurements were performed using an Iviumstat/Biologic SP150e electrochemical workstation in a three-electrode electrochemical cell in a de-aerated 0.1 M Na₂SO₄ solution (pH = 6.8), where Ag/AgCl (sat. KCl) was used as the reference electrode, a graphite rod was used as the auxiliary electrode, and the photocatalyst-modified fluorine-doped-tin-oxide (FTO) glass obtained by electrophoretic deposition was used as the working electrode. The transient photocurrent responses were measured at 1.23 V vs. RHE with chopping light illumination at a constant interval of 10 s. The electrochemical impedance spectra were recorded at 1.23 V vs. RHE in the frequency range of 1000 Hz–0.01 Hz with an amplitude of 0.05 V. Mott–Schottky (M–S) measurements were performed in 0.1 M Na₂SO₄ at a frequency of 1000 Hz. TRPL measurements were carried out using a Horiba DeltaPro Time-Correlated Single Photon Counting (TCSPC) Lifetime Fluorimeter with a 296 nm delta diode as the excitation source. The electron spin resonance (ESR) spectra were recorded using a JEOL with 5,5-dimethyl-1-pyrroline-*N*-oxide (DMPO) as a spin-trapping agent to detect the carbon-centred free radicals. A microwave frequency of 9.44 GHz and a microwave power of 0.995 mW were used. The sample was collected after 20 min to record the ESR spectra. For the solid samples, the spectra were recorded at room temperature.

Photocatalytic studies

Photocatalytic H₂ evolution. The photocatalytic tests were carried out in custom-build reactors using a custom-designed white LED as the light source. The temperature of the reactor was maintained using a water bath and the reactors was cooled with fans to avoid heating of the reactor. Prior to hydrogen evolution studies, CN and CN-*x*P catalysts were modified with Pt *via* photodeposition. Briefly, a known amount of photocatalyst was dispersed in 15 mL of water consisting of 10 vol% of methanol. To this, a required amount of H₂PtCl₆, corresponding to 1.5 wt% of Pt, was added and the mixture was sonicated for 20 min and de-aerated by purging with high pure Ar for 30 min. Subsequently, the mixture was illuminated for 1 h to photoreduce Pt-ions to Pt. After that, the sample was centrifuged, washed and kept for drying at 60 °C for 12 h before further use. Hydrogen evolution studies were carried out by suspending known amounts (10 mg) of CN and CN-*x*P in water containing 10 vol% TEOA that acts as the SED. Prior to the measurements, the reactor was deaerated by purging high-purity Ar for 30 min. The amount of H₂ produced was quantified by collecting 0.2 mL of gas from the headspace of the reactor at regular intervals of time using air tight Hamilton syringes. The collected samples were injected into a gas chromatograph (Agilent 8890) equipped with a TCD detector. The carrier gas used was Ar at a flow rate of 0.5 mL min⁻¹. A Mol-Sieve 5A 80/100 SS (9Ft 2 mm) packed column was used. The temperature of the column was maintained at 40 °C, and the inlet and detector temperatures were kept at 150 °C. The



apparent quantum efficiency (AQE) was determined in the presence of TEOA using LEDs that emit 452, 517, and 636 nm of light (Fig. S16†). The intensity of light was measured using a spectroradiometer. The AQE was determined using the following equation:

$$\text{AQE (\%)} = \frac{2 \times n_{\text{H}_2}}{n_{\text{photons}}} \times 100 = \frac{2 \times M \times N_A \times h \times c}{P \times S \times t \times \lambda_{\text{incident}}} \quad (6)$$

where n_{H_2} is the number of electrons transferred in H_2 evolution, n_{photons} is the total number of incident photons, M is the amount of H_2 in moles, N_A is the Avogadro number, h is Planck's constant, c is the speed of light, S is the irradiation area, P is the intensity of irradiation light, t is the photoreaction time and λ is the wavelength of incident light.

Photocatalytic co-production of H_2 and benzaldehyde

Studies on the co-production of H_2 and benzaldehyde were conducted in deaerated aqueous solutions of 20 mM BA. The amount of H_2 produced was measured by GC as mentioned above. The oxidation product formed during photocatalysis was characterized by ^1H NMR (Bruker Avance Neo 500 MHz). The NMR sample consists of 350 μL of the reaction mixture collected after a certain time of illumination from the photocatalyst, 100 μL D_2O and 100 μL of 1 mM maleic acid (internal standard). The NMR signal integrals were obtained using the Bruker Topspin software. The concentration of the species was calculated using the following formula:^{43b}

$$C_x = \frac{I_x}{I_{\text{std}}} \frac{N_{\text{std}}}{N_x} C_{\text{std}} \quad (7)$$

where I_x is the integral of the species, namely BADH (δ 9.8 ppm) or BA (δ 7.4–7.2 ppm), I_{std} is the integral of the internal standard (δ 6.2 ppm), N_{std} is the number of protons of internal standard, N_x is the number of protons of BA or BADH and C_{std} is the concentration of the internal standard.

The number of moles of the species was calculated using eqn (3):

$$n = \frac{C_x \times V_{\text{NMR tube}} \times V_{\text{R}}}{V_{\text{sample}}} \quad (8)$$

where C_x is the concentration of the species determined from eqn (2), $V_{\text{NMR tube}}$ is the total volume (sample + internal standard + D_2O) present in the tube, V_{R} is the total volume of the reaction mixture taken in reactor (10 mL in the present case), and V_{sample} is the volume of the sample aliquot from the reactor taken for analysis. For mass spectral studies, the reaction mixture collected after illumination of a known period of time was separated from the photocatalyst and was extracted in CH_2Cl_2 . Briefly, the catalyst was separated from the reaction mixture by filtering it through a 0.22 μm Nylon syringe filter and extracted with CH_2Cl_2 . The mixture extracted in CH_2Cl_2 is injected into GC equipped with an MS detector (HP-5 fused silica capillary column, 0.25 m \times 0.25 mm \times 30 m). The liquid chromatography (LC) measurements were carried out using an LC 1290 infinity (Model 6460 Triple Quadrupole) with a Zorbax Eclipse plus C18 rapid resolution column HD: 2.1 \times 50 mm, 1.8 microns. The mobile phase used for LC measurements was 70 :

30, that is 0.1% formic acid in water and acetonitrile. The column temperature was kept at 30 $^\circ\text{C}$.

DFT calculations

Periodic density functional theory (DFT) studies were conducted using the planewave based *ab initio* code, Vienna *Ab initio* Simulation Package (VASP).^{44–46} The interactions between the core and valence electrons were treated using the projector augmented wave (PAW) potentials.⁴⁷ The exchange-correlation energy density functional of Perdew–Burke–Ernzerhof along with Grimme's D3 semi-empirical method (PBE-D3) was used.^{48,49} Since the electronic band gaps from pure DFT functionals are known to be underestimated, a more accurate hybrid DFT functional by Heyd–Scuseria–Ernzerhof (HSE06)⁵⁰ was used to calculate the electronic band structure. Plane-wave basis-sets with a kinetic energy cut-off of 520 eV were used to expand the Kohn–Sham wave functions of the valence electrons. The graphical software VESTA was used to generate all the reported structures.⁵¹ The adsorption energies (ΔE) of different reaction intermediates over the surface were calculated as follows:

$$\Delta E(\text{int}) = E(\text{sur-int}) - [E(\text{sur}) + E(\text{int})] \quad (9)$$

where $E(\text{sur-int})$, $E(\text{sur})$ and $E(\text{int})$ represent the energies of intermediate (int) adsorbed surface, pure surface and reaction intermediate respectively. The energies of all the molecular systems were calculated in a cubic cell of 20 \AA length. The formation energy of different P-doped systems at both the N- and C-sites in $\text{g-C}_3\text{N}_4$ was calculated as follows:

$$\Delta E_{\text{F}} = E(\text{P@X-C}_3\text{N}_4) - E(\text{C}_3\text{N}_4) - E(\text{X}) \quad (10)$$

where $E(\text{P@X-C}_3\text{N}_4)$, $E(\text{C}_3\text{N}_4)$, and $E(\text{X})$ represent the energies of P doped at the X-site in $\text{g-C}_3\text{N}_4$, pure $\text{g-C}_3\text{N}_4$ and element X respectively. The energies of C, N and P were calculated from their standard states (graphite for C, N_2 gas for N and black phosphorus for P) as follows:

$$E(\text{X}) = \left[\frac{E(\text{X}_n)}{n} - E(\text{X}_{\text{iso}}) \right] \quad (11)$$

where $E(\text{X}_n)$ and $E(\text{X}_{\text{iso}})$ represent the energy of the molecule/material with n atoms in its standard state and the energy of a single atom in a large cubic cell respectively.

Author contributions

ADG carried out experimental work, data interpretation, write-up; SP assisted in synthesis and data interpretation; SK performed DFT calculations and write-up related to DFT; KV conceived the idea, overall supervision, manuscript writing and editing. All the authors have given approval for the final version.

Conflicts of interest

There are no conflicts to declare.



Acknowledgements

KV is indebted to SERB-DST (SRG/2020/000719) for financial support. ADG and SP are grateful to SERB-DST and BITS Pilani, K Birla Goa campus for research fellowship respectively. We gratefully acknowledge CSIF, BITS Pilani, K K Birla Goa campus for providing XRD, FESEM, Raman, NMR and LC-MS facilities. Central analytical laboratory, BITS Pilani, Hyderabad is acknowledged for XPS facility. Horiba India Technical Centre (HITC), Indian Institute of Science (IISc) is acknowledged for helping us with TRPL measurements. SAIF of IIT Bombay is acknowledged for ESR (ESR-JEOL) facility. We thank Mr Laxman Raikar and Prof. Halan Prakash, Department of Chemistry, BITS Pilani, K K Birla Goa campus for assisting us in measuring the intensity of LEDs used in the present study.

Notes and references

- 1 S. Chen, T. Takata and K. Domen, *Nat. Rev. Mater.*, 2017, **2**, 1–17.
- 2 J. H. Kim, D. Hansora, P. Sharma, J. W. Jang and J. S. Lee, *Chem. Soc. Rev.*, 2019, **48**, 1908–1971.
- 3 X. B. Li, C. H. Tung and L. Z. Wu, *Nat. Rev. Chem*, 2018, **2**, 160–173.
- 4 M. Y. Qi, M. Conte, M. Anpo, Z. R. Tang and Y. J. Xu, *Chem. Rev.*, 2021, **121**, 13051–13085.
- 5 L. I. Granone, F. Sieland, N. Zheng, R. Dillert and D. W. Bahnemann, *Green Chem.*, 2018, **20**, 11691192.
- 6 K. A. Davis, S. Yoo, E. W. Shuler, B. D. Sherman, S. Lee and G. Leem, *Nano Convergence*, 2021, **8**, 6.
- 7 R. Marotta, I. Di Somma, D. Spasiano, R. Andreozzi and V. Caprio, *Chem. Eng. J.*, 2011, **172**, 243–249.
- 8 S. Kampouri and K. C. Stylianou, *ACS Catal.*, 2019, **9**, 4247–4270.
- 9 J. Barrio, M. Volokh and M. Shalom, *J. Mater. Chem. A*, 2020, **8**, 11075.
- 10 (a) X. Wang, K. Maeda, A. Thomas, K. Takanabe, G. Xin, J. M. Carlsson, K. Domen and M. Antonietti, *Nat. Mater.*, 2009, **8**, 76–80; (b) O. Savateev and J. Zhuang, *ChemPhotoChem*, 2024, **8**, e202300306; (c) N. Karjule, R. S. Phatake, S. Barzilai, B. Mondal, A. Azoulay, A. I. Shames, M. Volokh, J. Albero, H. Garcia and M. Shalom, *J. Mater. Chem. A*, 2022, **10**, 16585–16594; (d) Q. Chen, C. Ning, J. Fang, B. Ping, G. Li, L. Kong, J. Chen, Z. Sun, J. Wang, Q. Ruan, X. Niu and L. Tao, *Chem. Eng. J.*, 2024, **487**, 150581; (e) M. Thangamuthu, Q. Ruan, P. O. Ohemeng, B. Luo, D. Jing, R. Godin and J. Tang, *Chem. Rev.*, 2022, **122**, 11778–11829; (f) Q. Ruan, M. K. Bayazit, V. Kiran, J. Xie, Y. Wang and J. Tang, *Chem. Commun.*, 2019, **55**, 7191–7194; (g) R. C. Sahoo, H. Lu, D. Garg, Z. Yin and H. S. S. R. Matte, *Carbon*, 2022, **192**, 101–108.
- 11 Y. Li, X. Li, H. Zhang and Q. Xiang, *Nanoscale Horiz.*, 2020, **5**, 765.
- 12 (a) L. Zhou, H. Zhang, H. Sun, S. Liu, M. O. Tade, S. Wang and W. Jin, *Catal. Sci. Technol.*, 2016, **6**, 7002–7023; (b) Y. Wang, M. K. Bayazit, S. J. A. Moniz, Q. Ruan, C. C. Lau, N. Martsinovich and J. Tang, *Energy Environ. Sci.*, 2017, **10**, 1643–1651.
- 13 H. Bin Fang, X. H. Zhang, J. Wu, N. Li, Y. Z. Zheng and X. Tao, *Appl. Catal., B*, 2018, **225**, 397–405.
- 14 X. Ma, Y. Lv, J. Xu, Y. Liu, R. Zhang and Y. Zhu, *J. Phys. Chem. C*, 2012, **116**, 23485–23493.
- 15 W. Wang, L. Du, R. Xia, R. Liang, T. Zhou, H. K. Lee, Z. Yan, H. Luo, C. Shang, D. L. Phillips and Z. Guo, *Energy Environ. Sci.*, 2023, **16**, 460–472.
- 16 J. Ran, T. Y. Ma, G. Gao, X. W. Du and S. Z. Qiao, *Energy Environ. Sci.*, 2015, **8**, 3708–3717.
- 17 B. Li, Y. Si, Q. Fang, Y. Shi, W. Q. Huang, W. Hu, A. Pan, X. Fan and G. F. Huang, *Nano-Micro Lett.*, 2020, **12**, 52.
- 18 Y. Wang, F. Silveri, M. K. Bayazit, Q. Ruan, Y. Li, J. Xie, C. R. A. Catlow and J. Tang, *Adv. Energy Mater.*, 2018, **24**, 1801084.
- 19 Q. Zhang, P. Chen, L. Chen, M. Wu, X. Dai, P. Xing, H. Lin, L. Zhao and Y. He, *J. Colloid Interface Sci.*, 2020, **568**, 117–129.
- 20 J. Zhang, W. Zhang, L. Yue, X. Hu, H. Lin, L. Zhao and Y. He, *Appl. Surf. Sci.*, 2022, **592**, 153337.
- 21 P. Chen, L. Chen, S. Ge, W. Zhang, M. Wu, P. Xing, T. B. Rotamond, H. Lin, Y. Wu and Y. He, *Int. J. Hydrogen Energy*, 2020, **45**, 14354–14367.
- 22 Y. Zhou, L. Zhang, J. Liu, X. Fan, B. Wang, M. Wang, W. Ren, J. Wang, M. Li and J. Shi, *J. Mater. Chem. A*, 2015, **3**, 3862–3867.
- 23 L. Zhang, X. Chen, J. Guan, Y. Jiang, T. Hou and X. Mu, *Mater. Res. Bull.*, 2013, **48**, 3485–3491.
- 24 H. Yang, Y. Zhou, Y. Wang, S. Hu, B. Wang, Q. Liao, H. Li, J. Bao, G. Ge and S. Jia, *J. Mater. Chem. A*, 2018, **6**, 16485–16494.
- 25 Y. P. Zhu, T. Z. Ren and Z. Y. Yuan, *ACS Appl. Mater. Interfaces*, 2015, **7**, 16850–16856.
- 26 L. Jiang, X. Yuan, Y. Pan, J. Liang, G. Zeng, Z. Wu and H. Wang, *Appl. Catal., B*, 2017, **217**, 388–406.
- 27 M. Wu, J. Zhang, B. bei He, H. wen Wang, R. Wang and Y. sheng Gong, *Appl. Catal., B*, 2019, **241**, 159–166.
- 28 M. Wen, N. Yang, J. Wang, D. Liu, W. Zhang, S. Bian, H. Huang, X. He, X. Wang, S. Ramakrishna, P. K. Chu, S. Yang and X. F. Yu, *ACS Appl. Mater. Interfaces*, 2021, **13**, 50988–50995.
- 29 F. Zhang, J. Li, H. Wang, Y. Li, Y. Liu, Q. Qian, X. Jin, X. Wang, J. Zhang and G. Zhang, *Appl. Catal., B*, 2020, **269**, 118772.
- 30 H. F. Ye, R. Shi, X. Yang, W. F. Fu and Y. Chen, *Appl. Catal., B*, 2018, **233**, 70–79.
- 31 H. Wang, X. Zhang, J. Xie, J. Zhang, P. Ma, B. Pan and Y. Xie, *Nanoscale*, 2015, **7**, 5152–5156.
- 32 J. Zou, Y. Yu, K. Qiao, S. Wu, W. Yan, S. Cheng, N. Jiang and J. Wang, *J. Mater. Sci.*, 2020, **55**, 13618–13633.
- 33 C. Yang, S. Zhang, Y. Huang, K. Lv, S. Fang, X. Wu, Q. Li and J. Fan, *Appl. Surf. Sci.*, 2020, **505**, 144654.
- 34 S. Zhao, Y. Liu, Y. Wang, J. Fang, Y. Qi, Y. Zhou, X. Bu and S. Zhuo, *J. Colloid Interface Sci.*, 2022, **616**, 152–162.
- 35 Y. Zhang, T. Mori, J. Ye and M. Antonietti, *J. Am. Chem. Soc.*, 2010, **132**, 6294–6295.



- 36 M. Alejandra Quintana, R. R. Solís, M. Ángeles Martín-Lara, G. Blázquez, F. Mónica Calero and M. J. Muñoz-Batista, *Sep. Purif. Technol.*, 2022, **298**, 121613.
- 37 G. Li, Y. Wu, M. Wang, W. Zhou, X. Liu, Z. Zhu, X. Song and P. Huo, *ACS Appl. Nano Mater.*, 2023, **6**, 14513–14526.
- 38 G. Zhang, G. Li, Z. A. Lan, L. Lin, A. Savateev, T. Heil, S. Zafeiratos, X. Wang and M. Antonietti, *Angew. Chem., Int. Ed.*, 2017, **56**, 13445–13449.
- 39 X. Chi, S. Tan, J. Song, F. Liu, Y. Tian, H. Yuan and R. Guan, *Catal. Lett.*, 2021, **151**, 3592–3602.
- 40 J. Wang, L. Guan, S. Yuan, J. Zhang, C. Zhao, X. Hu, B. Teng, Y. Wu and Y. He, *Sep. Purif. Technol.*, 2023, **314**, 123554.
- 41 J. Wang, H. Liang, C. Zhang, B. Jin and Y. Men, *Appl. Catal., B*, 2019, **256**, 117874.
- 42 X. Xiang Fang, L. B. Ma, K. Liang, S. J. Zhao, Y. F. Jiang, C. Ling, T. Zhao, T. Y. Cheang and A. W. Xu, *J. Mater. Chem. A*, 2019, **7**, 11506.
- 43 (a) C. Pulignani, C. A. Mesa, S. A. J. Hillman, T. Uekert, S. Gimenez, J. R. Durrant and E. Reisner, *Angew. Chem., Int. Ed.*, 2022, **134**, 202211587; (b) M. Abdinejad, S. Subramanian, M. K. Motlagh, M. Noroozifar, S. Duangdangchote, I. Neporozhni, D. Ripepi, D. Pinto, M. Li, K. Tang, J. Middelkoop, A. Urakawa, O. Voznyy, H. Kraatz and T. Burdyny, *Adv. Energy Mater.*, 2023, 2300402.
- 44 G. Kresse and J. Furthmüller, *Phys. Rev. B*, 1996, **54**, 11169–11186.
- 45 G. Kresse and J. Furthmüller, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 46 G. Kresse and D. Joubert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**, 1758–1775.
- 47 P. E. Blöchl, *Phys. Rev. B*, 1994, **50**, 17953–17979.
- 48 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 49 S. Grimme, S. Ehrlich and L. Goerigk, *J. Comput. Chem.*, 2011, **32**, 1456–1465.
- 50 A. V. Krukau, O. A. Vydrov, A. F. Izmaylov and G. E. Scuseria, *J. Chem. Phys.*, 2006, **125**, 224106.
- 51 K. Momma and F. Izumi, *J. Appl. Crystallogr.*, 2011, **44**, 1272–1276.

