

Cite this: *Digital Discovery*, 2025, 4, 943

ADEL: an automated drop-cast electrode setup for high-throughput screening of battery materials†

Maha Ismail,^{ab} Maria Angeles Cabañero,^{ad} Joseba Orive,^a Lakshmipriya Musuvadhi Babulal,^a Javier Garcia,^a Maria C. Morant-Miñana,^{id a} Jean-Luc Dauvergne,^a Francisco Bonilla,^a Iciar Monterrubio,^a Javier Carrasco,^{id ac} Amaia Saracibar^b and Marine Reynaud^{id *a}

Screening electrode materials in conventional battery research is time-consuming due to the lengthy and intricate preparation process, where multiple parameters directly influence electrochemical performance. In this work, we present ADEL, an affordable module for the Automated preparation of high-loading Drop-cast Electrodes, integrated within MAITENA, a Materials Acceleration and Innovation platform for ENergy Applications. The process consists of two main steps: (i) the automated preparation of electrode slurries and (ii) the drop-casting of these slurries onto aluminum foils using a pipetting robot, followed by drying under a halogen lamp. ADEL enables the preparation of 48 electrodes per day, allowing for the screening of up to 24 distinct active materials and/or electrode formulations. We demonstrate the method's repeatability using various commercial and lab-synthesized battery materials in different cell configurations, consistently achieving results with less than 3% relative standard deviation. As such, ADEL provides reliable, high-quality datasets for fast screening of battery materials, significantly accelerating research and development efforts.

Received 28th November 2024
Accepted 24th February 2025

DOI: 10.1039/d4dd00381k

rsc.li/digitaldiscovery

Introduction

Energy storage needs have dramatically increased driven by the growing demands in transportation and grid-scale applications. Next-generation batteries are expected to provide high energy capabilities, excellent safety measures, extended cycle life, significant sustainability, and cost-effectiveness.^{1,2} In this context, accelerating research efforts to explore new systems and chemistries is essential to meet the increasing demands and successfully tackle the challenges posed by the global energy transition.^{3,4}

The adoption of high-throughput methods, which refer to a workflow of running multiple analogous experiments simultaneously,⁵ is acknowledged as a crucial step to accelerate the discovery and optimization of new materials. These methods include computational methods,⁶⁻⁹ automated synthesis for combinatorial experimentation,¹⁰⁻¹⁵ and advanced characterization,¹⁶⁻¹⁸ and have been applied to the development

of materials for various energy-related fields such as batteries, electrocatalysis, photocatalysis, photovoltaics, and fuel cells.^{5,19-22} Moreover, a growing trend toward integrating these approaches with data-driven methods and artificial intelligence aims to elaborate “closed-loop” methodologies – usually implemented in Self-Driving Laboratories (SDLs) or Materials Acceleration Platforms (MAPs) – where feedback from previous trials is used to dynamically adjust and optimize the research process.²³⁻²⁹ Generating extensive and high-quality datasets is of primary importance for developing reliable predictive models, enabling efficient data analysis, and enhancing decision-making processes.³⁰ In this regard, automated setups enhance data reliability by minimizing human error due to manual intervention, ensuring consistency under experimental conditions, and enhancing reproducibility across trials.²⁰

The overall electrochemical performance of batteries depends on the critical properties of their components and their compatibility, which includes electrode active materials, electrolytes, conductive additives, binders, and current collectors.⁵ Significant work has recently been performed to develop high-throughput testing for liquid electrolytes.³⁰⁻³⁵ However, screening electrode materials constitutes a greater challenge, as optimizing the electrode manufacturing process involves multiple steps and interdependent variables, resulting in greater complexity and time consumption.^{36,37} The process requires preparing a well-mixed slurry made of an active material, a binder and a conductive material homogeneously

^aCentre for Cooperative Research on Alternative Energies (CIC EnergiGUNE), Basque Research and Technology Alliance (BRTA), Alava Technology Park, Albert Einstein 48, 01510 Vitoria-Gasteiz, Spain. E-mail: mreynaud@cicenergigune.com

^bPhysical Chemistry Department, Pharmacy Faculty, Basque Country University (UPV/EHU), 01006 Vitoria-Gasteiz, Alava, Spain

^cIKERBASQUE, Basque Foundation for Science, Plaza Euskadi 5, 48009 Bilbao, Spain

^dFEV Iberia, Calle Cardenal Gardoqui 1, 48008 Bilbao, Spain

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d4dd00381k>



dispersed in a solvent, which is then cast onto a current collector and dried. Each step presents key variables that must be controlled, such as the mixing time and viscosity, conditioned by the solid content, particle size distribution and morphology, as well as the electrode drying protocol. In addition, the electrode formulation and preparation process define the final electrode's porosity and microstructure, which directly impacts the electrochemical performance of the battery in terms of energy density and cyclability.^{36,38} Therefore, establishing a systematic method for automated lab-scale electrode preparation is essential to facilitate reliable and fast screening of electrode materials within accelerated battery material development workflows.

Various combinatorial and high-throughput methods have been proposed to overcome the time limitation of electrode preparation starting from the early 21st century. Sputtering techniques have been widely used to investigate thin film electrodes using multicell setups.^{39–45} McCalla *et al.*⁴⁴ adopted a multicell setup, developed by Dahn's group^{41,42} for thin films, to investigate 64 powder cathode materials synthesized through high-throughput coprecipitation employing precise drop-casting of the slurry onto adhered aluminum foil on a printed circuit board. This methodology requires high precision to get good reproducibility and avoid cross-contamination during the assembly of the cell. It led to specific capacities exhibiting a standard deviation of approximately 7.5% across the samples. Another approach involving micro-dispensing and high-speed jetting was investigated for different electrode materials and SiO₂-based separators with feature sizes above *ca.* 0.6–1 mm.⁴⁶ Although these studies are effective for understanding thin film batteries, they do not necessarily reflect the representative bulk active material behavior in commercial batteries.^{44,45}

While moving beyond thin-film and microscale electrodes appears necessary, studies focusing on testing representative bulk electrode materials remain scarce. Furthermore, they are largely limited to specific multicell designs and have not been extended to common lab-scale cell configurations such as coin cells, Swagelok cells, or tailor-made *operando* set-ups.^{15,44,47} Moreover, lab-scale electrodes are typically prepared and tested with low mass loading (ranging from 2 to 4 mg cm⁻²),⁴⁸ far from the requirements of commercial batteries, which demand high mass loading to achieve high energy density. Therefore, it is desirable to increase the mass loading of newly tested electrodes to evaluate them under conditions closer to those of actual batteries.⁴⁸

Here, we present an affordable automated module for high-throughput electrode preparation, named ADEL (Automated Drop-cast Electrodes setup), which is integrated into the self-driving lab MAITENA (Materials Acceleration and Innovation platform for ENergy Applications) at CIC EnergiGUNE. This setup enables the production of reproducible, high-quality electrode mass loadings exceeding those achieved with conventional lab-scale methods. It serves as an alternative to doctor blade casting and punching for electrode testing in various lab cell configurations. The reproducibility and suitability of the method are demonstrated using several state-of-the-art Li-ion cathode materials, and its practical applicability in MAPs is

illustrated through a case study screening multiple lithium-rich layered oxides as cathode active materials.

Experimental setup

The typical process for electrode preparation consists of casting a slurry of active materials and additives onto a current collector. Automating this procedure can be efficiently achieved through a pipetting workflow executed by liquid-handling robots. The primary challenge in this process lies in managing the pipetting of viscous electrode slurries, which requires thorough optimization to ensure reproducibility. The Ot-2 robot (Opentrons, USA) was chosen as an affordable and efficient solution that is ready to use, easy to operate through protocol designers, and capable of handling high-viscosity liquids.^{49,50} Considering the limited space available for each deck slot in this compact system, the robot was equipped with a MIXdrive 24 MTP multi-position stirring plate (from MIXdrive 24P), selected as the most suitable option to enable the simultaneous preparation of 24 electrode slurries.

Fig. 1 presents a sketch of ADEL, a setup developed to automate the electrode preparation. A photo and a video of the setup are provided in Fig. SI-1(a), Section S1 (ESI).[†] The module was built around an Ot-2 robot, featuring two pipettes of 300 and 1000 μ l ([A] in main elements). This robot was equipped with several 3D-printed custom key components (see Section S2, ESI[†]), including a multi-vial holder to accommodate the initial binder solutions and solvents [B] (Fig. SI-2[†]), a holder specifically designed to hold up to 24 test tubes filled with solid powders [C] on top of the multi-position stirring plate [D] (Fig. SI-3[†]), and up to 6 holders to fix aluminum foils [E] (Fig. SI-4(d)[†]). A magnet (III in other accessories) was used as a pipette tip to recover the magnetic bars from the tubes when required. In addition, the module includes a halogen lamp [F] to facilitate the drying process, as described later.

The process for preparing electrodes using this setup is described as follows. The initial liquid mixtures [B] are prepared by mixing a defined amount of a binder (*e.g.* polyvinylidene fluoride, PVDF) and a solvent (*e.g.* *N*-methylpyrrolidone, NMP). Weighing the solid powders (active material and carbon source) in the test tubes [C] is carried out separately for each component using an automatic solid dispenser [G] (Mettler Balance XPR106DUHQ, see Fig. SI-1(b) in Section S1, ESI[†]), placed later in the tube holder of the multi-position stirring plate [D]. The automated electrode preparation process initiates with the pipetting of 500 μ l from the mixture [B] onto 100 mg of solid powders within each tube [C]. Following this, two hours of mixing at 900 rpm ensures a well-mixed and homogeneous slurry consistency, ensuring the desired viscosity, which will be discussed later. It is to be noted that the mixing duration may vary depending on the materials' properties (*e.g.* particle morphology and agglomerates) and, in some cases, may require adjustments based on slurry consistency. However, in this work, the mixing duration was fixed at two hours for all materials studied. The second step involves pipetting 100 μ l of the prepared slurry onto the aluminum foils to form each electrode, allowing pipetting 4 electrodes from each tube. The pipetting



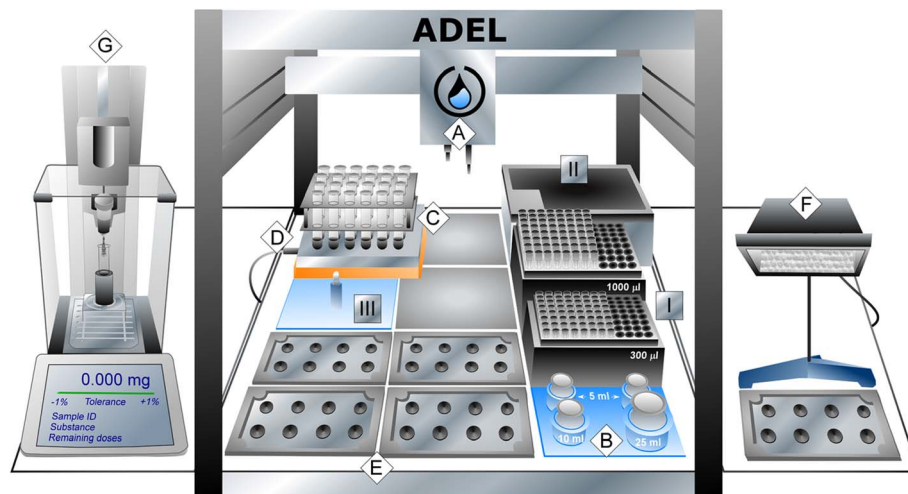


Fig. 1 Sketch of the ADEL setup for automated battery electrode preparation for fast material screening. Main elements of the sketch: (A) Ot-2 pipettes; (B) initial binder solutions and solvents; (C) test tubes containing active materials and carbon powder; (D) MIXdrive 24 multi-position stirring plate; (E) Al foil holders; (F) halogen lamp; (G) automatic solid dispenser. Other accessories in the sketch: (I) pipette tips; (II) waste container; (III) magnet tip.

parameters used for the aspiration/dispensing steps are provided in Table SI-1† and will be described in the pipetting optimization section. The six custom-printed holders [E] enable the preparation of 48 electrodes per run (*i.e.* at least 2 electrodes from each slurry). The entire process, including automated solid weighting (half an hour), slurry mixing (2 hours), and finally electrode dispensing (half an hour), requires a total of 3 hours per run.

The initial drying of the electrodes is carried out at room temperature for 3 to 4 hours to preserve their shape. Then, they are dried under a halogen lamp [F] for typically 3 hours. Afterward, the electrodes are punched with an automated 12 mm puncher (Nogami) and weighted individually. Finally, the electrodes are dried in a Buchi oven at 120 °C to eliminate any moisture before being transferred to a glove box for cell assembly.

While the drop-casting method is fully automated *via* the Ot-2 robot, tube transfers from the automated balance to the Ot-2 and Al foil holder transfers to the halogen lamp were done here manually. These brief interventions at the beginning and the end of the process do not require continuous researcher presence, but also serve as key checkpoints for verifying sample quality. Further automation using a robotic arm would be possible given the flexibility of the ADEL setup, but is beyond the scope of this study. Overall, the ADEL setup allows for the screening of up to 24 distinct active materials, with minimal human intervention required for initial and final transfers.

Results and discussion

In order to optimize the automated electrode preparation process, $\text{LiNi}_{0.5}\text{Mn}_{0.3}\text{Co}_{0.2}\text{O}_2$ (NMC-532, MTI EQ-Lib-LNCM523) was chosen as a model commercial cathode active material to prepare a series of electrodes using a slurry formulation of 84 : 8 : 8 wt% (active material : carbon : binder). The optimization of

the process includes viscosity evaluation, pipetting and drying optimization steps, as presented in Fig. 2. Given the viscous nature of the slurry, optimizing the aspiration and dispensing protocols was crucial to ensure the reproducibility of the results. This optimization seeks to prevent distorted results caused by potential volume variations of the pipetted liquid, which can arise from bubbles, air gaps, and any residual fluid in the pipette tip during the pipetting process. To this end, two viscous mixtures of glycerin and water, GW75 (75% wt glycerin and 25% wt water) and GW90 (90% wt glycerin and 10% wt water), were first employed to optimize the pipetting parameters and mimic the viscosity of the slurry, with particular emphasis on regulating the flow rate, as depicted in Table SI-1.† Supplementary features of the Ot-2 robot were applied to improve pipetting accuracy, including adjustments of the tip position, timing delays, and utilization of the blowout function (see Table SI-1†). The viscosity values of the NMC slurry as a function of the shear rate fall between the viscosities of the two GW mixtures, as shown in Fig. 2(a) (details of the viscosity test are provided in Section S3, ESI†). In contrast to the Newtonian behavior of the GW mixtures, the viscosity of the NMC slurry decreases with the shear rate, which is indicative of the shear thinning behavior of non-Newtonian slurries. The viscosity values of the NMC slurry are close to 10^{-2} Pa s across the entire shear rate range and these low values³⁷ can be explained by the low solid content (17 wt%) of the dispersion. This is notably lower than the solid content typically used on the lab scale (25 to 40 wt%) and manufacturing scale (50 to 65 wt%) for slurry preparation.⁵¹ This low solid content is required for the drop-casting electrode preparation method employed in this process. To ensure successful pipetting, we adopted the pipetting parameters used for the most viscous sample (GW90) at the pipetting shear rate (1375 s^{-1}), considering the viscosity difference as a safety margin.



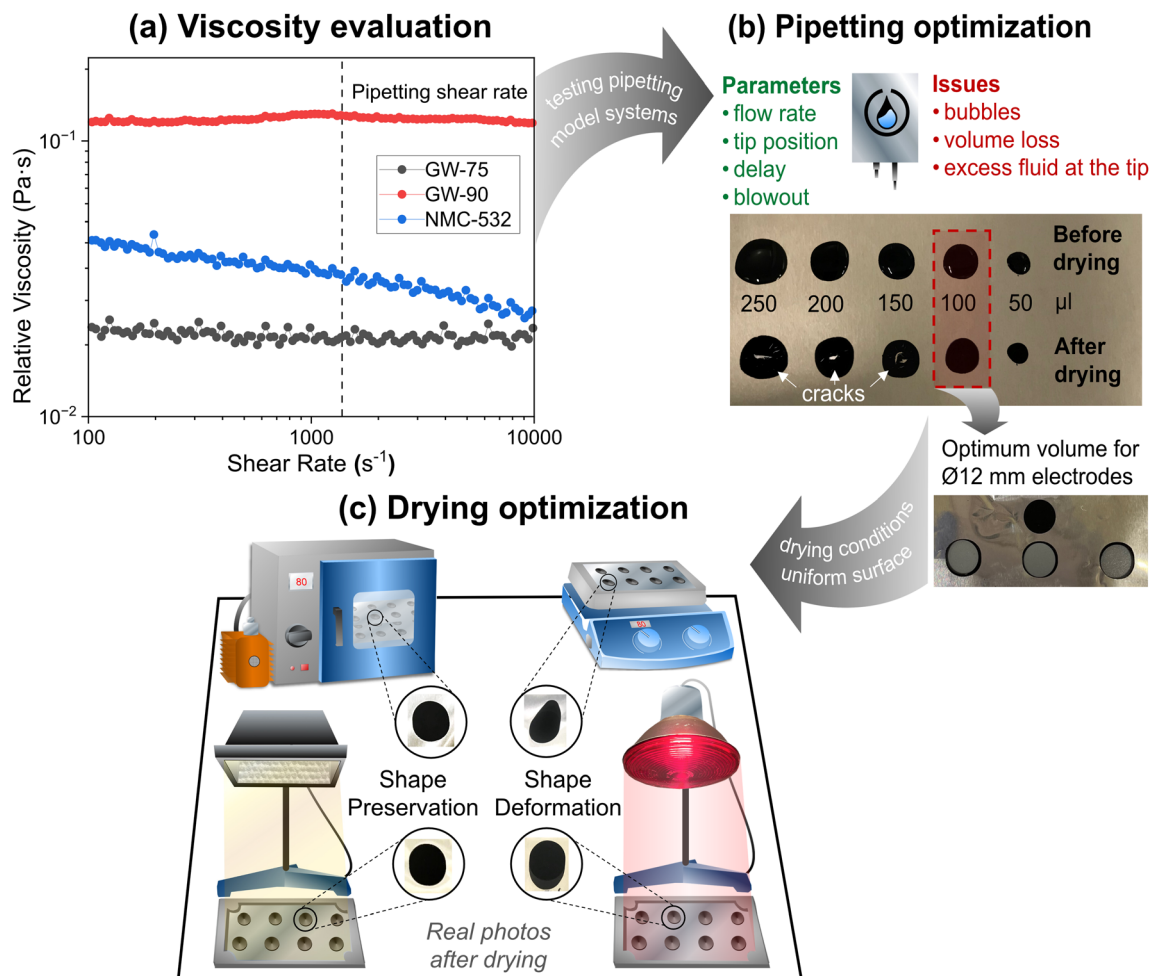


Fig. 2 Optimization workflow followed to establish the automated electrode preparation process. (a) The viscosity evaluation chart shows the variation of the viscosity of the samples in the interested shear rate zone. (b) The pipetting optimization identifies the important parameters and issues to be controlled during the aspiration and dispensing process and the optimized volume of the electrodes. (c) The drying optimization represents the different drying methods tested during the whole process: vacuum oven, heating plate, infrared lamp, and halogen lamp.

Next, the volume of pipetting was adjusted to ensure the formation of a well-defined 12 mm electrode after drying, as illustrated in Fig. 2(b). To determine the optimal conditions, the dropped volume was varied between 250 and 50 μl . When using volumes higher than 150 μl , the resulting electrodes presented cracks after drying, attributed to the higher concentration of solids in the center of the electrode as the result of the spread of the NMP solvent towards the edges. The best results were eventually obtained using a volume of 100 μl , consistently yielding reproducible 12 mm diameter electrodes free of cracks. Note that, similarly to the doctor blade method, which requires the application of vacuum to maintain the aluminum foil, the shape and quality of electrodes produced in the automated module also depend on the flatness of the Al foils. Therefore, several Al foil tailor-made holder designs were investigated (see Fig. SI-4 in Section S2, ESI†). The final design incorporated neodymium magnets positioned at the corners of the holder, providing optimal fixation of the Al foil. This approach also facilitated the easy handling and removal of the foils after electrode preparation, while reducing both costs and space required for the final setup.

Drying the electrodes right away after pipetting was initially attempted; however, it resulted in crack formation. Therefore, the electrodes were first preconditioned by leaving them at room temperature for 3 to 4 hours to slowly evaporate the solvent. Then, four different drying methods – vacuum oven, heating plate, infrared lamp, and halogen lamp – were evaluated (Fig. 2(c)). The drying process (for the heating plate, infrared lamp, and halogen lamp) was monitored on an hourly basis to assess when it appeared to be relatively completed. Once the electrode was observed to be free of solvent, an additional hour of drying was applied to ensure thorough and complete drying. The standard drying method at 80 °C for 8 hours under vacuum yielded crack-free dried electrodes (Fig. 2(c)). To accelerate the drying step, the use of a heating plate was evaluated, but this resulted in the deformation of the drop shapes (see Fig. 2(c)) and inhomogeneous electrode loading after cutting.

Alternatively, the previously reported use of a round infrared (IR) lamp (PHILIPS PAR38 IR)⁵² was tested. The homogeneity of this illumination was assessed using an infrared camera to map the apparent temperatures of the whole Al foil surface during



the drying process (details of the illumination test are provided in Section S4, ESI†). The recorded data presented in Fig. 3(a) showed that this configuration led to non-uniform heating with variations that could exceed 10 °C across the Al foil. While a rectangular IR lamp should help to mitigate these special variations in illumination, the use of a halogen lamp is a more affordable alternative. Likewise, the temperature gradients provided by the illumination of a halogen lamp (RS PRO Floodlight, 400 W) are presented in Fig. 3(b), showing that the use of this lamp substantially improved the temperature distribution on the electrode area, with a deviation of only 3 °C between the center and the borders. The implementation of this halogen lamp for electrode drying, which to the best of our knowledge has not been reported elsewhere, proved to effectively dry the electrodes within a remarkably short period of 3 hours. This cost-effective and easily accessible solution was subsequently implemented. Regardless of the drying process, the electrochemical performance of all electrodes consistently yields results closely similar to those obtained with the standard lab method (*i.e.* doctor blade coating) (see Fig. SI-5 in Section S4 and Fig. SI-6 in Section S5, ESI†). Therefore, these results show that using the halogen lamp for drying electrodes offers significant advantages in terms of time savings and affordability, making it an attractive electrode drying option to adopt in any laboratory.

Following the optimization process described above, the electrochemical performance of the NMC-532 electrodes

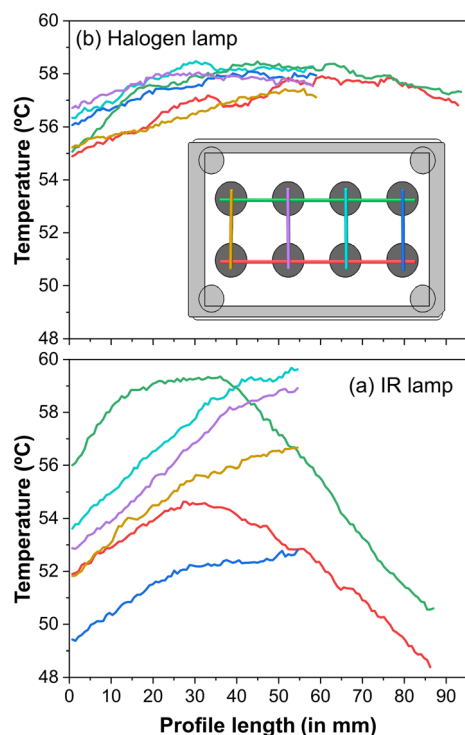


Fig. 3 Apparent temperature profiles of the Al foil along different directions recorded using an IR camera, (a) when heating using an IR lamp and (b) when heating using a halogen lamp. The x-axis in mm represents the length of the profiles taken on the electrode surface, as shown in the inset.

produced by this protocol was first compared to that of electrodes produced using the standard lab-scale doctor blade method, showing very similar results with an average capacity of 168.05 mA h g⁻¹ for the doctor blade method and 169.80 mA h g⁻¹ for the ADEL method (Fig. SI-6 in Sections S5 and S8 for electrochemical testing details, ESI†). To assess the global electrodes' homogeneity, Micro-Computed Tomography (micro-CT) was performed on sections of two of the NMC-532 electrodes, enabling a 3D reconstruction of the electrodes' surface and thickness (see experimental details in ESI, Section S5†). Both reconstructions (shown in Fig. SI-7(a) and (b)†) indicate a consistent thickness across the electrodes, with the ADEL electrode measuring 109.9 ± 9.8 μm and the doctor blade electrode 73.8 ± 5.8 μm. To complement the micro-CT findings and achieve higher resolution, scanning electron microscopy (SEM) analysis was performed on cross-sections of the same samples, prepared using ion milling (IM) techniques (see experimental details in ESI, Section S5†). This approach enabled high-resolution visualization of microstructural features within individual particles, achieving nanometric spatial resolution. By integrating micro-CT for macroscopic thickness measurements with SEM for high-resolution microstructural analysis, a multiscale characterization strategy was established, providing a comprehensive assessment of the electrode's structural integrity and material distribution. As shown in Fig. 4(a) and (b) from the IM cross-section SEM images, a significantly greater thickness – nearly double – was observed for the electrode prepared using the ADEL method, confirming that the drop-cast electrodes exhibited substantially higher mass loading. In addition, both images show a relatively uniform distribution of the active material, conductive additive, and binder across the electrode thickness, regardless of the casting method. In terms of surface homogeneity, some defects were detected on the drop-cast electrode, while the doctor blade electrode appears relatively smooth. However, these defects are minor and within an acceptable range, as supported by the electrochemical results, which will be discussed in detail later. Overall, the comparison of these two methods shows that the automated protocol is faster and more efficient, minimizing material waste by converting the entire slurry into a defined number of drop-cast electrodes that are later assembled in the electrochemical cells. In addition, despite the low solid content of the slurry, the ADEL setup produces high-loading electrodes, closer to actual battery requirements.

The reproducibility of the slurry preparation was assessed using a series of six NMC-532 drop-cast electrodes from three distinct slurries (*i.e.* two electrodes per slurry). Fig. 5(a) shows the specific capacity of the cells after cycling at C/20, C/10, C/5, and 1C, respectively (the corresponding data are presented in Table SI-2 and Section S5, ESI†). The six cells exhibit remarkably comparable capacities at each C rate and are in good agreement with the reported data for NMC-532,⁵³ demonstrating the high reproducibility of the prepared slurries.

To go further in the statistical data and calculate the standard deviation of the method, another series of ten drop-cast electrodes of NMC-532 was prepared. They were assembled into coin cells and subjected to the same cycling method as



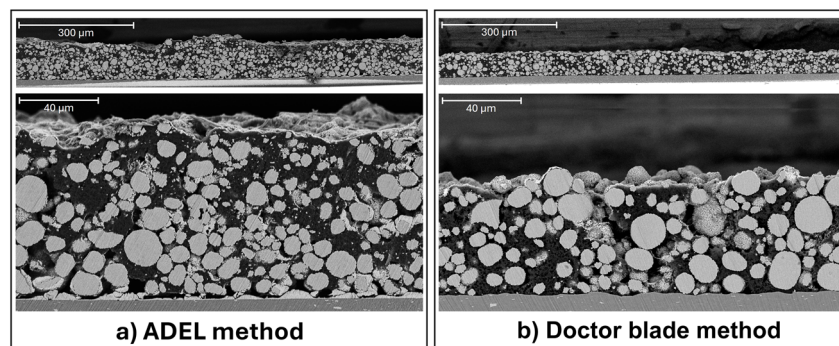


Fig. 4 SEM cross-section images of NMC-532 electrodes prepared by (a) the drop-cast ADEL method and (b) the standard lab-scale doctor blade method.

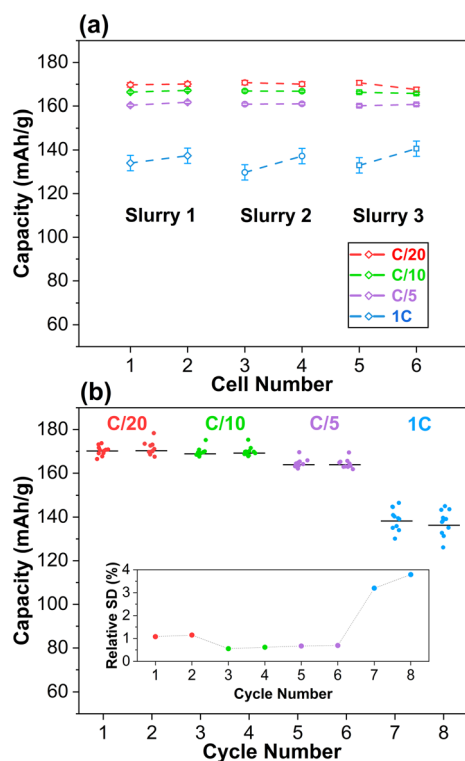


Fig. 5 (a) Slurry reproducibility test illustrated by the NMC-532 specific capacity at different C rates (C/20, C/10, C/5, and 1C) of two cycled cells for two electrodes prepared from three different slurries. (b) Electrode reproducibility test showing the specific capacity of the NMC-532 electrodes prepared using the ADEL module, illustrating the dispersion of the results at various C rates of C/20, C/10, C/5, and 1C, respectively. The inset represents the relative standard deviation (SD) of the results at each C rate.

described above. Fig. 5(b) displays the distribution of the specific capacity obtained for each cell at different C-rates, together with their respective relative standard deviation (data presented in Table SI-3, Section S5, ESI†). The average mass loading over 10 electrodes was $12.12 \pm 0.72 \text{ mg cm}^{-2}$ (Table SI-3†), which represents a 6% mass variation, indicating a remarkable uniformity of the weight obtained after electrode pipetting. The average capacity obtained at C/20 is

$170.4 \text{ mA h g}^{-1}$ with a relative standard deviation of 1.1% between electrodes, while cycling at C/10 and C/5 provides slightly lower average discharge capacities but maintains the relative standard deviation below 1% at both rates. Increasing the C-rate up to 1C leads to an average capacity of $137.6 \text{ mA h g}^{-1}$ with a slightly higher relative standard deviation of 3.5%. Despite this increase, the standard deviation obtained at 1C remains very low, especially considering that the electrodes were not calendared. Generally, numerous factors influence the electrochemical performance of electrode materials beyond electrode preparation,³⁶ including the electrolyte used for the electrode assessment³⁸ as well as cell-to-cell variations due to cell assembly.⁵⁴ Taking all these parameters into account, the automated protocol presented herein reflects significant reproducibility with a very small standard deviation, and to the best of our knowledge, is not reported elsewhere at these C-rates using other automatization setups.^{44,55}

After demonstrating the suitability of ADEL for producing NMC layered oxide electrodes, other state-of-the-art positive electrode materials (*i.e.* spinel and olivine materials) were tested to validate the versatility of the module toward the main families of Li-ion cathode active materials. In this regard, three commercial electrode materials exhibiting diverse particle size distributions and morphologies were employed: (i) high-voltage spinel $\text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_4$ (LNMO, TBM-129 from TOPSOE) having spherical particles with an average particle size between 5 and 20 μm , (ii) carbon-coated LiFePO_4 (LFP/C, from Advanced Lithium Electrochemistry Co.), consisting of tabular particles with diameters ranging from 50 to 300 nm,⁵⁶ and (iii) spinel LiMn_2O_4 presenting nanoparticles with particle sizes below 500 nm (LMO, >99% Sigma-Aldrich). To complete this materials assessment, 5 V spinel $\text{LiFe}_{0.5}\text{Mn}_{1.5}\text{O}_4$ (LFMO) synthesized in our laboratory (see Section S7, ESI†) was also tested to verify that the ADEL automated drop-cast electrode method can be effectively applied to non-optimized materials, which may present larger particle size distribution, non-uniform morphologies, and composition variability. To further validate the reproducibility of the preparation of both the slurries and electrodes, two separate slurries were prepared for each material, and two electrodes were produced from each slurry. The same exact formulation of 84 : 8 : 8 wt% (active material : carbon : binder)



was used for all the selected materials. Fig. 6(a-1)–(d-1) show the charge and discharge curves of the first cycle of the four cells prepared for each material – LFP, LMO, LNMO, and LFMO (electrochemical testing details are provided in Section S8, ESI†). All cycling curves show the characteristic behavior of the tested materials. LFP/C (Fig. 6(a-1)) presents a flat plateau centered at 3.4 V vs. Li^+/Li^0 with an average capacity of $153.2 \pm 1.2 \text{ mA h g}^{-1}$ at C/20.⁵⁷ The voltage profile and capacity of $115.9 \pm 2.4 \text{ mA h g}^{-1}$ obtained for the LMO material (Fig. 6(b-1)) align well with the data reported in the literature.^{58,59} LNMO (Fig. 6(c-1)) exhibits the characteristic feature of the $\text{Mn}^{3+}/\text{Mn}^{4+}$ redox couple around 4.0 V Li^+/Li^0 in addition to the two high-voltage plateaus associated with $\text{Ni}^{2+}/\text{Ni}^{3+}$ and $\text{Ni}^{3+}/\text{Ni}^{4+}$ around 4.8 V vs. Li^+/Li^0 , resulting in an average capacity of $136.3 \pm 0.5 \text{ mA h g}^{-1}$ upon discharge, consistent with reported cycling data.⁶⁰ The electrochemical curves of LFMO presented in Fig. 6(d-1) exhibit the typical two high-voltage processes associated with the $\text{Mn}^{3+}/\text{Mn}^{4+}$ and $\text{Fe}^{3+}/\text{Fe}^{4+}$ redox couples, with an average capacity of $130.0 \text{ mA h g}^{-1} \pm 1.1 \text{ mA h g}^{-1}$, in line with

the best results reported in the literature.⁶¹ The low reversibility over the first charge/discharge cycle is not inherent to the electrode preparation but is attributed to electrolyte side reactions at high voltage and to the material itself. It is worth noting that none of the 16 electrodes tested exhibited higher polarization than what is typically observed with conventional electrode preparation. This is particularly relevant for characterizing high-voltage cathode materials such as LNMO or LFMO, which require cycling at voltages up to 5.2 V vs. Li^+/Li^0 . In such cases, any overpotential arising from the electrode preparation can dramatically hinder the material's performance. On the other hand, the evolution of the capacity over the first 6 cycles is also presented for all 16 cells in Fig. 6(a-2)–(d-2), showing high reproducibility with a variation of less than 5 mA h g^{-1} between the four cells of each material. A higher dispersion of the capacity results is observed at higher C-rates and in the LMO cells, where the nanometric size of the LMO particles may slightly affect the reproducibility of the slurry and

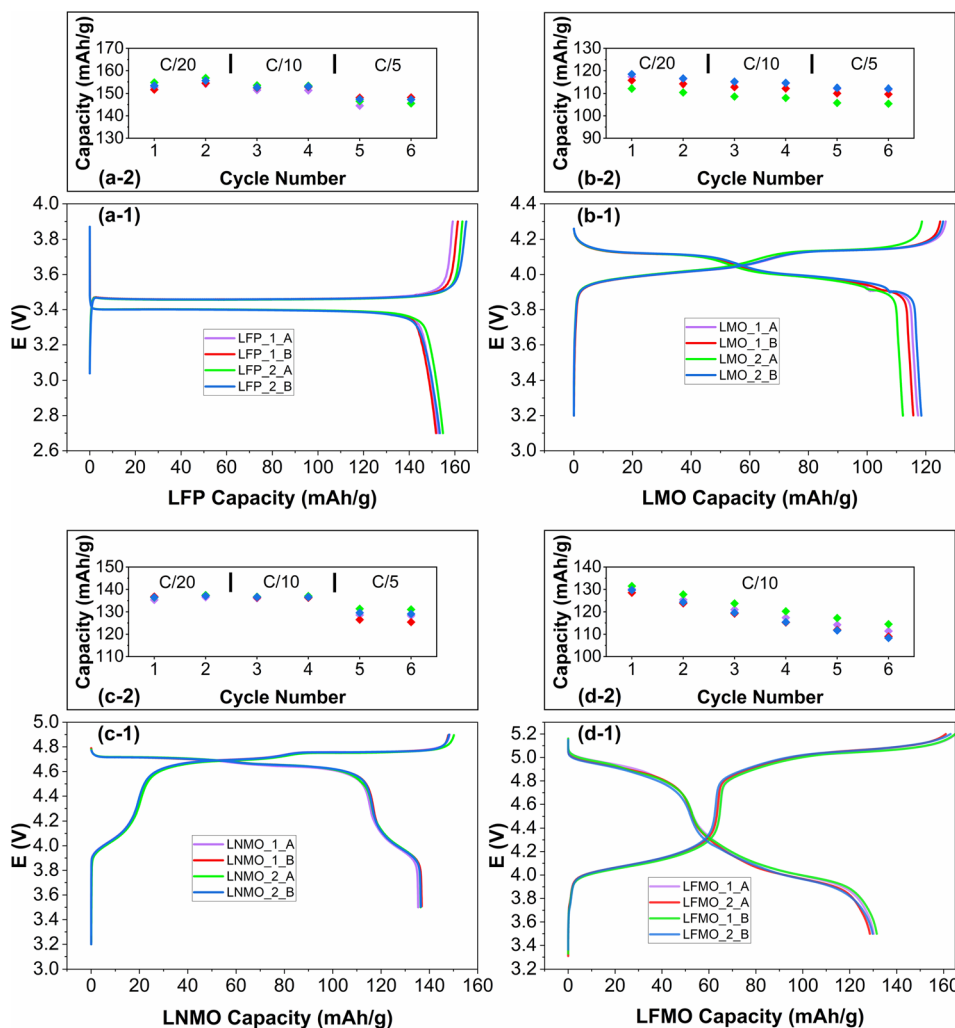


Fig. 6 Cycling profile of the four tested cells for (a-1) LFP electrodes at C/20, (b-1) LMO electrodes at C/20, (c-1) LNMO electrodes at C/20 and (d-1) LFMO electrodes at C/10. (a-2)–(d-2) The cyclability of the cells over the first six cycles at different C rates for LFP, LMO, LNMO, and LFMO materials, respectively.



electrode drop-casting,⁶² though it remains within a satisfactory range.

After having validated the automated ADEL method with several state-of-the-art cathode materials, it was employed to screen the electrochemical performance of a series of samples of Li-rich layered oxides. Lithium-rich materials represent a promising and challenging family of high-energy cathode materials, which offer extra capacity by involving anionic redox reactions. Low initial coulombic efficiencies, voltage fading due to structural transformations and limited rate capability in full cells are some of the challenges that can be tackled through the optimization of these materials. In this case study, we investigated the effects of varying Li content and annealing temperature on the coulombic efficiency and discharge capacity of lab-prepared $\text{Li}_{1+x}(\text{Mn,Ni})\text{O}_2$ (LMR) samples (see Section S7 for material synthesis, and Section S8 for electrochemical testing details in the ESI†). The reproducibility of the slurries and electrodes was firstly assessed considering a specific selected composition, $\text{Li}_{1.35}(\text{Mn,Ni})\text{O}_2$, by preparing three distinct slurries, with two electrodes per slurry. The discharge capacities of the formation cycles at C/20 and the subsequent two cycles at C/10 for the resulting six LMR electrodes are displayed in Fig. SI-8 (Section S6, ESI†). The obtained discharge capacity values are tabulated in Table SI-4.† Each LMR electrode displayed a similar discharge capacity with an average value of 272.05 mA

h g^{-1} at C/20 and a relative standard deviation of 1.0%. In the second and third cycles at C/10, the electrodes still deliver high discharge capacities around 250 mA h g^{-1} with a relative standard deviation below 2%. Then, a series of 20 different LMR samples were synthesized at five distinct temperatures with varying lithium content, defined as the lithium to transition metal ratio (Li/TM). Fig. 7 compares the discharge capacity and coulombic efficiency of the formation cycle at C/20 and the second cycle at C/10 obtained from all samples. Discharge capacities for two cells per sample are shown to illustrate the repeatability of the results. As expected, the electrodes exhibited relatively low initial coulombic efficiencies, around 70–80% for the samples with lower lithium contents. Additionally, a declining trend in efficiency was observed for Li/TM ratios above 1.55 (Fig. 7(a)). The coulombic efficiencies were stabilized around 95–98% for the second cycle, but the downward tendency is still evident for the more Li-rich compositions (Fig. 7(b)). Similar behavior was observed for the discharge capacity of the first two cycles (Fig. 7(c) and (d)). All in all, the fast screening of lithium-rich oxides with the ADEL module also enables the identification of candidates with outstanding parameters, such as the specific LMR samples with initial coulombic efficiencies above 90% and discharge capacities greater than 250 mA h g^{-1} (Fig. 7(a) and (c)).

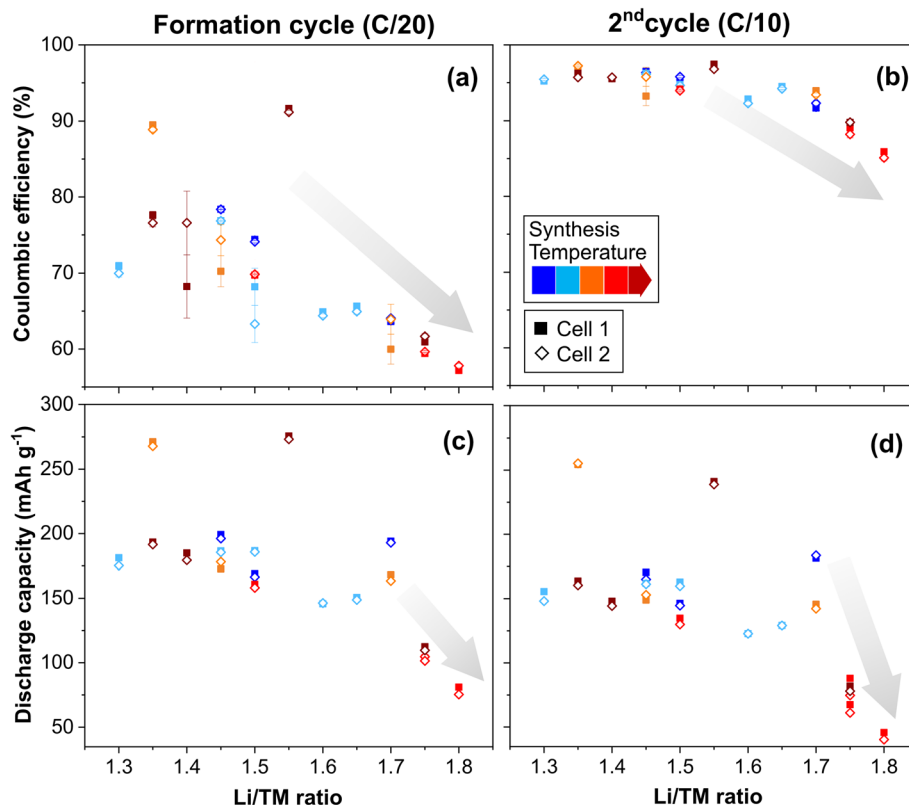


Fig. 7 (a and b) Coulombic efficiency and (c and d) discharge capacity of the formation cycle (C/20) and second cycle (C/10) for a wide range of LMR electrodes with the $\text{Li}_{1+x}(\text{Mn,Ni})\text{O}_2$ formula containing varying Li content (Li/TM ratio from 1.3 to 1.8). The color legend represents the annealing temperature of the material from lowest (blue) to highest (maroon). Note that the filled squares represent cell 1 and empty diamonds represent a replicate of the first one (cell 2) for each electrode slurry. The error bars are shown in the figure, which in some cases are smaller than the squares/diamonds themselves. TM: transition metal.



Conclusion

The automated setup ADEL demonstrates robust capabilities for high-throughput preparation of drop-cast electrodes. It enables the automated preparation of 48 high-loading electrodes in a single batch, allowing for the screening of up to 24 distinct electrode materials per day with minimal human intervention, all within a 10 hour workflow. This includes 3 hours for electrode drop-casting and 7 hours of complete drying. In comparison, it is estimated that the conventional doctor blade casting requires around 1 hour of significant human intervention for each electrode laminate,⁴⁵ which is followed by 8 hours of drying in a vacuum oven. This results in a total processing time of approximately 32 hours for producing 24 distinct electrodes. While the configuration presented herein allows for a maximum of four electrodes per slurry, it can be easily adapted to scale up the number of drop-cast electrodes per material.

In this study, the electrochemical characterization of 72 drop-cast electrodes at varying C-rates in different cell types (coin and Swagelok cells), representing 25 different commercial and lab-synthesized powder cathode active materials, has shown that the ADEL system produces highly reproducible results with a low relative standard deviation below 3% in all cases. While this work focused on cathode active material evaluation, the ADEL methodology can also be applied to screen different anode materials and/or various electrode formulations (*e.g.*, testing different binders or conductive materials and varying their content). In addition, the system is designed to be adaptable and compact, making it suitable for placement inside a glove box.

To the best of our knowledge, the ADEL setup is one of the most affordable and efficient methods for screening the electrochemical performance of over 100 distinct electrode slurries per week. As such, we believe that it constitutes a valuable module for battery-related Materials Accelerated Platforms (MAPs) and Self-Driving Labs (SDLs), which will significantly contribute to the acceleration of battery materials research.

Data availability

The data that support the findings of this study are openly available in the following GitHub repository: <https://github.com/Maha24cic/ADEL-CIC>. The corresponding DOI is the following: <https://doi.org/10.5281/zenodo.14905113>. This repository includes a video of the module, the 3D-printed component plans permitting to reproduce the setup, and all data associated with this work.

Author contributions

Maha Ismail: methodology, investigation, conceptualization, formal analysis, validation, visualization, writing – original draft, and writing – review & editing. Maria Angeles Cabañero: methodology, investigation, conceptualization, validation, and writing – review & editing. Joseba Orive: investigation, formal analysis, visualization, writing – original draft, and writing – review & editing. Lakshmi Priya Musuvadhi Babulal:

investigation, formal analysis, validation, and writing – original draft. Javier Garcia: software and investigation. Maria C. Morant-Miñana: investigation, formal analysis, and writing – review & editing. Jean-Luc Dauvergne: investigation, formal analysis, and writing – review & editing. Francisco Bonilla: formal analysis and writing – review & editing. Iciar Monterrubio: formal analysis and validation. Javier Carrasco: funding acquisition and writing – review & editing. Amaia Saracibar: supervision, conceptualization, and writing – review & editing. Marine Reynaud: methodology, supervision, conceptualization, funding acquisition, and writing – review & editing.

Conflicts of interest

The authors declare no competing financial interest.

Acknowledgements

This work was supported by the ION-SELF project [ref. PID2019-106519RB-I00] and the PhD grant PRE2020-092978, funded by MICIU/AEI/10.13039/501100011033 and “ESF Investing in your future” as well as the project SMART [ref. PID2022-140823OB-I00] funded by MICIU/AEI/10.13039/501100011033 and ERDF/EU. The authors are grateful to UMICORE for providing the carbonate precursors for LMR materials and to TOPSOE for providing the LNMO materials used for this investigation. The authors acknowledge Ainhoa Bustinza for providing the planes of the 3D designs, Mattia Felice for assisting in the GitHub repository and Luz Granados for performing the micro-CT measurements.

References

- 1 S. Duehnen, J. Betz, M. Kolek, R. Schmuck, M. Winter and T. Placke, Toward green battery cells: perspective on materials and technologies, *Small Methods*, 2020, 4(7), 2000039.
- 2 R. Schmuck, R. Wagner, G. Hörpel, T. Placke and M. Winter, Performance and cost of materials for lithium-based rechargeable automotive batteries, *Nat. Energy*, 2018, 3(4), 267–278.
- 3 J. Amici, P. Asinari, E. Ayerbe, P. Barboux, P. Bayle-Guillemaud, R. J. Behm, M. Berecibar, E. Berg, A. Bhowmik and S. Bodoardo, A roadmap for transforming research to invent the batteries of the future designed within the european large scale research initiative battery 2030+, *Adv. Energy Mater.*, 2022, 12(17), 2102785.
- 4 S. Stier, C. Kreisbeck, H. Ihssen, M. Albert Popp, J. Hauch, K. Malek, M. Reynaud, F. Goumans, J. Carlsson, I. Todorov, L. Gold, A. Räder, W. Wenzel, S. T. Bandesha, P. Jacques, F. Garcia-Moreno, O. Arcelus, P. Friederich, S. Clark, M. Maglione, A. Laukkanen, I. E. Castelli, M. Casas Cabanas, J. C. Carrasco, H. S. Stein, T. Vegge, S. Nakamae, M. Fabrizio and M. Kozdras, The Significance of Accelerated Discovery of Advanced Materials to address Societal Challenges, *Zenodo*, 2023, (1.0.0).
- 5 A. Benayad, D. Diddens, A. Heuer, A. N. Krishnamoorthy, M. Maiti, F. L. Cras, M. Legallais, F. Rahmanian, Y. Shin



- and H. Stein, High-throughput experimentation and computational freeway lanes for accelerated battery electrolyte and interface development research, *Adv. Energy Mater.*, 2022, **12**(17), 2102678.
- 6 S. Adams and R. P. Rao, High power lithium ion battery materials by computational design, *Phys. Status Solidi A*, 2011, **208**(8), 1746–1753.
 - 7 S. Adams and R. P. Rao, Understanding Ionic Conduction and Energy Storage Materials with Bond-Valence-Based Methods, in *Bond Valences*, ed. I. D. Brown and K. R. Poeppelmeier, Springer Berlin Heidelberg, Berlin, Heidelberg, 2014, pp. 129–159.
 - 8 N. A. Katcho, J. Carrete, M. Reynaud, G. Rousse, M. Casas-Cabanas, N. Mingo, J. Rodriguez-Carvajal and J. Carrasco, *J. Appl. Crystallogr.*, 2019, **52**(1), 148–157.
 - 9 A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner and G. Ceder, Commentary: The Materials Project: A Materials Genome Approach to Accelerating Materials Innovation, *APL Mater.*, 2013, **1**(1), 011002.
 - 10 T. Adhikari, A. Hebert, M. Adamic, J. Yao, K. Potts and E. McCalla, Development of high-throughput methods for sodium-ion battery cathodes, *ACS Comb. Sci.*, 2020, **22**(6), 311–318.
 - 11 P. Liu, B. Guo, T. An, H. Fang, G. Zhu, C. Jiang and X. Jiang, High throughput materials research and development for lithium ion batteries, *J. Materiomics*, 2017, **3**(3), 202–208.
 - 12 S. Jia, J. Counsell, M. Adamič, A. Jonderian and E. McCalla, High-throughput design of Na–Fe–Mn–O cathodes for Na-ion batteries, *J. Mater. Chem. A*, 2022, **10**(1), 251–265.
 - 13 M. Roberts and J. Owen, High-throughput method to study the effect of precursors and temperature, applied to the synthesis of $\text{LiNi}_{1/3}\text{Co}_{1/3}\text{Mn}_{1/3}\text{O}_2$ for lithium batteries, *ACS Comb. Sci.*, 2011, **13**(2), 126–134.
 - 14 Y. Yao, Z. Huang, T. Li, H. Wang, Y. Liu, H. S. Stein, Y. Mao, J. Gao, M. Jiao and Q. Dong, High-throughput, combinatorial synthesis of multimetallic nanoclusters, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**(12), 6316–6322.
 - 15 A. Jonderian, S. Jia, G. Yoon, V. T. Cozea, N. Z. Galabi, S. B. Ma and E. McCalla, Accelerated Development of High Voltage Li-Ion Cathodes, *Adv. Energy Mater.*, 2022, **12**(40), 2201704.
 - 16 K. Mader, F. Marone, C. Hintermüller, G. Mikuljan, A. Isenegger and M. Stampanoni, High-throughput full-automatic synchrotron-based tomographic microscopy, *J. Synchrotron Radiat.*, 2011, **18**(2), 117–124.
 - 17 J. Gregoire, D. Van Campen, C. Miller, R. Jones, S. Suram and A. Mehta, High-throughput synchrotron X-ray diffraction for combinatorial phase mapping, *J. Synchrotron Radiat.*, 2014, **21**(6), 1262–1268.
 - 18 Y. Lyu, Y. Liu, T. Cheng and B. Guo, High-throughput characterization methods for lithium batteries, *J. Materiomics*, 2017, **3**(3), 221–229.
 - 19 T. Muster, A. Trinchì, T. Markley, D. Lau, P. Martin, A. Bradbury, A. Bendavid and S. Dligatch, A review of high throughput and combinatorial electrochemistry, *Electrochim. Acta*, 2011, **56**(27), 9679–9699.
 - 20 G. Tom, S. P. Schmid, S. G. Baird, Y. Cao, K. Darvish, H. Hao, S. Lo, S. Pablo-García, E. M. Rajaonson and M. Skreta, Self-driving laboratories for chemistry and materials science, *Chem. Rev.*, 2024, **124**(16), 9633–9732.
 - 21 S. P. Stier, C. Kreisbeck, H. Ihssen, M. A. Popp, J. Hauch, K. Malek, M. Reynaud, T. Goumans, J. Carlsson and I. Todorov, Materials Acceleration Platforms (MAPs) Accelerating Materials Research and Development to Meet Urgent Societal Challenges, *Adv. Mater.*, 2024, 2407791.
 - 22 J. Abed, Y. Bai, D. E. Persaud, J. Kim, J. Witt, J. Hattrick-Simpers and E. H. Sargent, AMPERE: automated modular platform for expedited and reproducible electrochemical testing, *Digital Discovery*, 2024, **3**(11), 2265–2274.
 - 23 H. S. Stein and J. M. Gregoire, Progress and prospects for accelerating materials science with automated and autonomous workflows, *Chem. Sci.*, 2019, **10**(42), 9640–9649.
 - 24 H. S. Stein, A. Sanin, F. Rahmanian, B. Zhang, M. Vogler, J. K. Flowers, L. Fischer, S. Fuchs, N. Choudhary and L. Schroeder, From materials discovery to system optimization by integrating combinatorial electrochemistry and data science, *Curr. Opin. Electrochem.*, 2022, **35**, 101053.
 - 25 *BIG-MAP project*, <https://www.big-map.eu/>.
 - 26 A. Aspuru-Guzik and K. Persson, Materials Acceleration Platform: Accelerating Advanced Energy Materials Discovery by Integrating High-Throughput Methods and Artificial Intelligence, *Mission Innovation: Innovation Challenge* 6, 2018, <https://nrs.harvard.edu/urn-3:HUL.InstRepos:35164974>.
 - 27 M. M. Flores-Leonar, L. M. Mejía-Mendoza, A. Aguilar-Granda, B. Sanchez-Lengeling, H. Tribukait, C. Amador-Bedolla and A. Aspuru-Guzik, Materials acceleration platforms: on the way to autonomous experimentation, *Curr. Opin. Green Sustainable Chem.*, 2020, **25**, 100370.
 - 28 T. Lombardo, M. Duquesnoy, H. El-Bouysidy, F. Årén, A. Gallo-Bueno, P. B. Jørgensen, A. Bhowmik, A. Demortière, E. Ayerbe and F. Alcaide, Artificial intelligence applied to battery research: hype or reality?, *Chem. Rev.*, 2021, **122**(12), 10899–10969.
 - 29 M. Abolhasani and E. Kumacheva, The rise of self-driving labs in chemical and materials sciences, *Nat. Synth.*, 2023, 1–10.
 - 30 S. Matsuda, K. Nishioka and S. Nakanishi, High-throughput combinatorial screening of multi-component electrolyte additives to improve the performance of Li metal secondary batteries, *Sci. Rep.*, 2019, **9**(1), 6211.
 - 31 A. Dave, J. Mitchell, S. Burke, H. Lin, J. Whitacre and V. Viswanathan, Autonomous optimization of non-aqueous Li-ion battery electrolytes via robotic experimentation and machine learning coupling, *Nat. Commun.*, 2022, **13**(1), 5454.
 - 32 C. Cartier, Z. Feng, J. Faulk and D. Scherson, A combinatorial approach toward the discovery of electrolyte formulations for non-aqueous electrochemical energy storage devices, *ECS Electrochem. Lett.*, 2015, **4**(9), A110.
 - 33 J. T. Yik, L. Zhang, J. Sjölund, X. Hou, P. H. Svensson, K. Edström and E. J. Berg, Automated electrolyte formulation and coin cell assembly for high-throughput lithium-ion battery research, *Digital Discovery*, 2023, **2**(3), 799–808.



- 34 J. Noh, H. A. Doan, H. Job, L. A. Robertson, L. Zhang, R. S. Assary, K. Mueller, V. Murugesan and Y. Liang, An integrated high-throughput robotic platform and active learning approach for accelerated discovery of optimal electrolyte formulations, *Nat. Commun.*, 2024, **15**(1), 2757.
- 35 A. Dave, J. Mitchell, K. Kandasamy, H. Wang, S. Burke, B. Paria, B. Póczos, J. Whitacre and V. Viswanathan, Autonomous discovery of battery electrolytes with robotic experimentation and machine learning, *Cell Rep. Phys. Sci.*, 2020, **1**(12), 100264.
- 36 T. Marks, S. Trussler, A. Smith, D. Xiong and J. Dahn, A guide to Li-ion coin-cell electrode making for academic researchers, *J. Electrochem. Soc.*, 2010, **158**(1), A51.
- 37 W. B. Hawley and J. Li, Electrode manufacturing for lithium-ion batteries—analysis of current and next generation processing, *J. Energy Storage*, 2019, **25**, 100862.
- 38 F. Dai and M. Cai, Best practices in lithium battery cell preparation and evaluation, *Commun. Mater.*, 2022, **3**(1), 64.
- 39 J. Whitacre, W. West and B. Ratnakumar, A Combinatorial Study of $\text{Li}_y\text{Mn}_x\text{Ni}_{2-x}\text{O}_4$ Cathode Materials Using Microfabricated Solid-State Electrochemical Cells, *J. Electrochem. Soc.*, 2003, **150**(12), A1676.
- 40 A. Spong, G. Vitins, S. Guerin, B. Hayden, A. Russell and J. R. Owen, Combinatorial arrays and parallel screening for positive electrode discovery, *J. Power Sources*, 2003, **119**, 778–783.
- 41 M. Fleischauer, T. Hatchard, A. Bonakdarpour and J. R. Dahn, Combinatorial investigations of advanced Li-ion rechargeable battery electrode materials, *Meas. Sci. Technol.*, 2004, **16**(1), 212.
- 42 M. D. Fleischauer, T. D. Hatchard, G. P. Rockwell, J. M. Topple, S. Trussler, S. K. Jericho, M. H. Jericho and J. R. Dahn, Design and testing of a 64-channel combinatorial electrochemical cell, *J. Electrochem. Soc.*, 2003, **150**(11), A1465.
- 43 M. Al-Maghrabi, N. van der Bosch, R. Sanderson, D. Stevens, R. Dunlap and J. Dahn, A new design for a combinatorial electrochemical cell plate and the inherent irreversible capacity of lithiated silicon, *Electrochem. Solid-State Lett.*, 2011, **14**(4), A42.
- 44 K. P. Potts, E. Grignon and E. McCalla, Accelerated screening of high-energy lithium-ion battery cathodes, *ACS Appl. Energy Mater.*, 2019, **2**(12), 8388–8393.
- 45 E. McCalla, Semiautomated Experiments to Accelerate the Design of Advanced Battery Materials: Combining Speed, Low Cost, and Adaptability, *ACS Eng. Au*, 2023, **3**(6), 391–402.
- 46 R. Hahn, M. Ferch, K. Tribowski, N. A. Kyeremateng, K. Hoepfner, K. Marquardt, K.-D. Lang and W. Bock, High-throughput battery materials testing based on test cell arrays and dispense/jet printed electrodes, *Microsyst. Technol.*, 2019, **25**, 1137–1149.
- 47 K. Takada, K. Fujimoto, T. Sasaki and M. Watanabe, Combinatorial electrode array for high-throughput evaluation of combinatorial library for electrode materials, *Appl. Surf. Sci.*, 2004, **223**(1–3), 210–213.
- 48 C. Cao, F. Liang, W. Zhang, H. Liu, H. Liu, H. Zhang, J. Mao, Y. Zhang, Y. Feng, X. Yao, M. Ge and Y. Tang, Commercialization-Driven Electrodes Design for Lithium Batteries: Basic Guidance, Opportunities, and Perspectives, *Small*, 2021, **17**(43), 2102233.
- 49 K. Watson and A. Kanase, *Viscous Liquid Handling Automation using OpenTrons, OT-2*, <https://opentrons.com.cn/wp-content/uploads/2023/11/Viscous-Liquid-Handling-Automation-using-OpenTrons-OT-2.pdf>.
- 50 *How to handle viscous liquids in Protocol Designer*, <https://support.opentrons.com/s/article/How-to-handle-viscous-liquids-in-Protocol-Designer>.
- 51 L. Ouyang, Z. Wu, J. Wang, X. Qi, Q. Li, J. Wang and S. Lu, The effect of solid content on the rheological properties and microstructures of a Li-ion battery cathode slurry, *RSC Adv.*, 2020, **10**(33), 19360–19370.
- 52 A. Altvater, T. Heckmann, J. C. Eser, S. Spiegel, P. Scharfer and W. Schabel, (Near-) Infrared Drying of Lithium-Ion Battery Electrodes: Influence of Energy Input on Process Speed and Electrode Adhesion, *Energy Technol.*, 2023, **11**(5), 2200785.
- 53 S. J. An, J. Li, Z. Du, C. Daniel and D. L. Wood III, Fast formation cycling for lithium ion batteries, *J. Power Sources*, 2017, **342**, 846–852.
- 54 P. Dechent, S. Greenbank, F. Hildenbrand, S. Jbabdi, D. U. Sauer and D. A. Howey, Estimation of Li-Ion Degradation Test Sample Sizes Required to Understand Cell-to-Cell Variability, *Batteries Supercaps*, 2021, **4**(12), 1821–1829.
- 55 B. Zhang, L. Merker, A. Sanin and H. S. Stein, Robotic cell assembly to accelerate battery research, *Digital Discovery*, 2022, **1**(6), 755–762.
- 56 C. Berlanga, I. a. Monterrubio, M. Armand, T. Rojo, M. Galceran and M. Casas-Cabanas, Engineering, cost-effective synthesis of triphylite- NaFePO_4 cathode: a zero-waste process, *ACS Sustainable Chem. Eng.*, 2019, **8**(2), 725–730.
- 57 H. Zheng, L. Chai, X. Song and V. Battaglia, Electrochemical cycling behavior of LiFePO_4 cathode charged with different upper voltage limits, *Electrochim. Acta*, 2012, **62**, 256–262.
- 58 S. Megahed and B. Scrosati, Lithium-ion rechargeable batteries, *J. Power Sources*, 1994, **51**(1–2), 79–104.
- 59 N. S. Choi, Z. Chen, S. A. Freunberger, X. Ji, Y. K. Sun, K. Amine, G. Yushin, L. F. Nazar, J. Cho and P. G. Bruce, Challenges facing lithium batteries and electrical double-layer capacitors, *Angew. Chem., Int. Ed.*, 2012, **51**(40), 9994–10024.
- 60 M. Fehse, N. Etxebarria, L. Otaegui, M. Cabello, S. Martín-Fuentes, M. A. Cabañero, I. Monterrubio, C. F. Elkjær, O. Fabelo and N. A. Enkubari, Influence of transition-metal order on the reaction mechanism of LNMO cathode spinel: an *operando* X-ray absorption spectroscopy study, *Chem. Mater.*, 2022, **34**(14), 6529–6540.
- 61 A. Eftekhari, Electrochemical performance and cyclability of $\text{LiFe}_{0.5}\text{Mn}_{1.5}\text{O}_4$ as a 5 V cathode material for lithium batteries, *J. Power Sources*, 2003, **124**(1), 182–190.
- 62 K.-S. Chen, R. Xu, N. S. Luu, E. B. Secor, K. Hamamoto, Q. Li, S. Kim, V. K. Sangwan, I. Balla and L. M. Guiney, Comprehensive enhancement of nanostructured lithium-ion battery cathode materials *via* conformal graphene dispersion, *Nano Lett.*, 2017, **17**(4), 2539–2546.

