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REVIEW ARTICLE

Electrolyte engineering promoting high-specific-energy lithium batteries in low-temperature environments

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Lithium batteries have become one of the preferred power sources for various external devices due to their high energy density, mature industrial infrastruture, and diverse applications. However, the increasing demands for enhanced functionality, broader operating conditions, and increased robustness in next-generation devices highlight a critical challenge: the poor performance of lithium batteries in low-temperature environments. At low temperatures, slow lithium-ion diffusion and charge transfer dynamics, closely linked to the electrolyte, significantly hinder battery performance. The electrolyte, which facilitates ionic transport and mediates various interfacial reactions between electrodes, is pivotal in addressing these limitations. This review identifies five key factors limiting battery performance in low-temperature environments and outlines comprehensive optimization strategies to address them. These include the engineering regulation of individual electrolyte components and the compatibility coordination among various components. We thoroughly elucidate the mechanisms behind existing optimization strategies and propose future development directions and prospects for advancing low-temperature lithium battery electrolytes. By intergrating rapidly evolving interdisciplinary strategies, this discussion aims to overcome the current limitations and pave the way for the next generation of high-performance lithium batteries for low-temperature environments.

the modernization of electrical equipment, the demand for low-

Broader context

The modernization of electrical systems is driving an escalating demand for high-specific-energy batteries. Lithium-based systems (Li-ion/Li-metal), recognized for their high energy density and mature industrial infrastructure, dominate electrochemical energy storage. However, conventional lithium batteries operate efficiently only under moderate conditions and fail in extreme environments (e.g., high plateaus, deep sea, polar regions, and space). Prolonged exposure to low temperatures causes drastic capacity loss or functional failure, primarily due to sluggish ion transport and interfacial instability. As the core components governing ion conduction and electrode-electrolyte interfacial dynamics, electrolytes are pivotal in addressing these cryogenic challenges. The key factors that limit the performance of lithium batteries in lowtemperature environments, along with comprehensive optimization strategies to address these factors and the underlying mechanisms—with particular emphasis on the compatibility and coordination among various components of the electrolyte should be thoroughly discussed.

1. Introduction

With the rapid advancement of military-civilian integration and

temperature lithium batteries is rising, especially in extreme environments such as plateaus, deep seas, polar regions, high altitudes, and outer space (Fig. 1a). The demand for lowtemperature lithium batteries is also increasing for applications in new energy vehicles, 5/6G base stations, military drones, and navigation systems. 1,2 Traditional lithium-ion batteries cannot meet the energy storage and release demands in these lowtemperature environments.3

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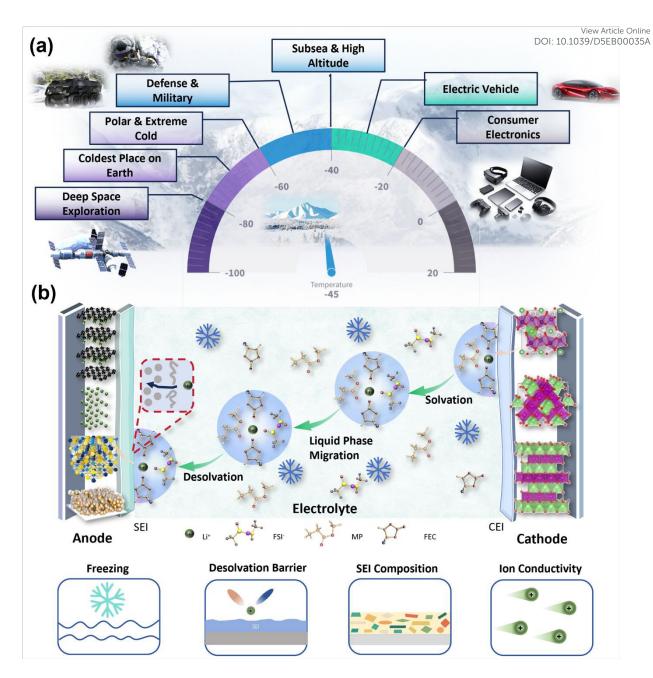


Fig. 1 a) Potential applications of LBs in low-temperature environments. b) Schematic representation of the discharge mechanism of LBs under low-temperature conditions.

In the military, low-temperature lithium batteries are widely used across all branches, including land, sea, air, space, and communications. These batteries are critical for various military applications, including large bases, small infantry units, aerial and space equipment, underwater devices, and unmanned combat systems. In the civilian sector, the development of the Internet of Vehicles (IoV), upgrades in consumer products, and the construction of new infrastructure have led to increased demand for low-cost, low-temperature lithium batteries for new energy transportation, logistics, and communications. In northern regions with extreme weather, such as harsh winters and high altitudes, the range of new energy vehicles is significantly reduced. Thus, the development of low-

temperature lithium-ion batteries that can operate in cold environments is essential.

Temperature affects mass transport and charge transfer processes during the charging and discharging of lithium batteries. At low temperatures, the reduced kinetics of lithium ions are the primary factor limiting electrochemical performance. Figure 1b shows the charging process of lithium batteries at low temperatures. ⁴⁻⁶ As shown in the figure, during charging and discharging, lithium ion migration proceeds through the following steps: 1) transport of dissolved lithium ions in the liquid phase; 2) solvation/desolvation; 3) migration at the phase interface (including the solid electrolyte interface, SEI, and cathode/electrolyte interface, CEI); 4) migration within

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the solid phase. At low temperatures, the ionic conductivity of the electrolyte decreases, hindering the solvation/desolvation of lithium ions. The migration rate of lithium ions within the electrode material and at the solid-liquid interface is significantly reduced, leading to substantial degradation in battery performance. Additionally, the anode surface becomes more susceptible to lithium plating and dendrite growth, which can cause battery failure and pose safety risks.^{7,8}

The electrolyte, as a key medium bridging the anode and cathode electrodes and facilitating ion transport, significantly impacts the battery's low-temperature performance. At low temperatures, the ionic conductivity of the electrolyte decreases, charge transfer kinetics slow down, and the migration rate of lithium ions at the SEI and within the electrode decreases (Fig. 1b). These factors collectively contribute to the degradation of battery performance in cold conditions. Low temperatures also reduce the ionic conductivity of the electrolyte, increase the dissociation energy barrier, and destabilize the SEI, significantly decreasing battery capacity.9 To address these challenges, solvents or additives with low freezing points and viscosities, combined with lithium salts that have high dissociation efficiency, can effectively lower the dissociation energy barrier, enhance ionic conductivity, and modify the solvation structure of lithium ions, thus optimizing battery performance in cold conditions. Despite recent advances, further efforts are needed to extend the lowtemperature operating range and improve battery energy density, as current results are limited to 140 Wh kg⁻¹ at -60°C (Fig 2). 10,11 This review summarizes key factors contributing to the performance degradation of high-energy batteries in lowtemperature conditions, analyzes the underlying mechanisms, and discusses recent advancements in low-temperature electrolyte engineering. The article focuses on components such as lithium salts, solvents, and additives, and outlines future research directions for electrolytes in high-energy, lowtemperature batteries.

2. Low Temperature Challenges

2.1 Thermodynamic and Kinetic Constraints on Low Temperature Redox Reactions.

As it shown in the Arrhenius equation, temperature is a critical factor influencing both the feasibility of a chemical reaction and its rate.

$$k = Ae^{-Ea/RT} \tag{1}$$

This equation shows that the rate of an electrochemical reaction decreases exponentially with temperature reduction, leading to considerable energy barriers within the reaction pathway. It can be inferred from these principles that the performance of batteries at low temperatures is susceptible to instability, thereby affecting the discharge specific capacity and voltage range during cycling. This poses a significant challenge for effective battery management. Within a certain temperature range, the kinetic coefficient demonstrates a strong temperature dependence. At low temperatures, ion

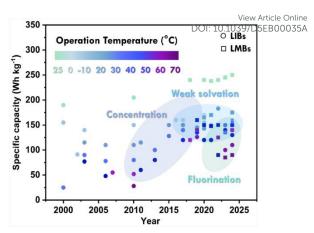


Fig. 2 Development on energy density for Lithium Batteries operated at low temperatures. The solid and dashed lines represent the energy density of lithium-ion batteries (LIBs) and lithium metal batteries (LMBs) from -70°C to 25°C, respectively

diffusion is hindered, leading to increased electrode polarization, and the cutoff voltage is rapidly reached, reducing output capacity.12

The thermodynamic and kinetic properties of redox reactions at both electrodes are adversely affected by reduced temperatures, severely limiting reaction rates and lithium-ion diffusion. This bottleneck reduces the battery's capacity and energy efficiency. Engineering electrolyte additives that modulate redox kinetics at low temperatures can help alleviate these constraints, allowing for more efficient electron and ion transfer, thus improving overall cell efficiency.

2.2 Drastic Decline in Ionic Conductivity

Temperature also significantly influences the physical and chemical properties of the electrolyte, as demonstrated by the Stokes-Einstein equation and dielectric phenomena.:

$$D = \frac{kT}{6\pi m} \tag{2}$$

$$q = \frac{|ZiZj|e^2}{8\pi\epsilon\epsilon_0 kT}$$
 (3)

where D (diffusion coefficient), η (medium's viscosity), γ radius), q (critical (solvation distance for formation), Z (ion valency), ε and ε_0 (dielectric constants of the medium and vacuum), respectively, and k (Boltzmann 's constant). Furthermore, ionic conductivity, σ , is given by:

$$\sigma = \sum_{i} n_{i} \mu_{i} Z_{i} e \tag{4}$$

where n_i (number of free ions), μ_i (ionic mobility), and Z_i (ion valency). Equations (2)-(4) show that at low temperatures, increased viscosity reduces ion mobility, and solvation shells form close ion pairs with counterions, impeding ion migration. The dielectric constant, ϵ , can either promote or inhibit ion dissociation, impacting the number of free ions, n_i, and thus

At sub-zero temperatures, conventional lithium battery electrolytes exhibit a marked decline in ionic conductivity. This phenomenon is attributed to increased electrolyte viscosity and restricted ion mobility, hindering efficient lithium-ion transport across the cell. As a result, charge/discharge rates and energy

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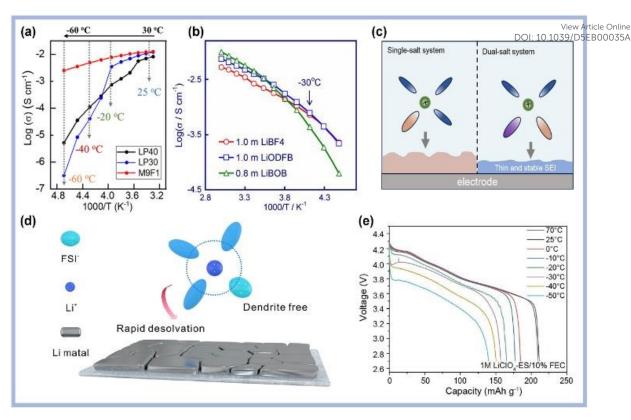


Fig. 3 a) lonic conductivities of different electrolytes measured at various temperatures. Reproduced from Ref. [18]. Copyright 2021, American Chemical Society. b) lonic conductivity of 1.0 m Li BF4,1.0 m LiODFB, and 0.8 m LiBOB in PC/EC/EMC (1:1:3 by wt). Reproduced from Ref. [19]. Copyright 2022, Elsevier. c) Schematic of mechanism of dual-salt. d) Schematic diagram of the relationship between Li deposition morphology and solvating power of solvent at low temperature. Reproduced from Ref. [23]. Copyright 2022, Wiley-VCH. e) Cycling performance of NCM811||Li cells with 1 M LiClO4-ES/10% FEC at different temperature. Reproduced from Ref. [24]. Copyright 2024, Wiley-VCH.

output are significantly impaired.¹³ Addressing this issue requires precisely engineered electrolyte formulations with low-viscosity solvents and optimized ion solvation mechanisms to sustain conductivity at low temperatures.

2.3 Formation of High-Resistance SEI.

As it shown in the Arrhenius equation, each process's kinetics declines exponentially with temperature. Low temperatures promote the formation of a dense, resistive SEI layer, largely due to incomplete SEI stabilization. This resistive SEI layer inhibits lithium-ion diffusion, leading to increased interfacial impedance. To mitigate this, low-temperature-compatible SEI-forming additives are essential, promoting a flexible, conductive SEI layer that can enhance ion permeability and minimize impedance under harsh conditions.

2.4 Electrolyte Crystallization and Phase Instability.

Conventional organic solvents in liquid electrolytes are prone to crystallization or phase separation at low temperatures, disrupting the continuity of the ion transport pathways. Such phase instability not only compromises ionic conductivity but also increases internal cell resistance. Developing supercooled or low-melting-point solvents can stabilize electrolyte phases, thereby preventing phase separation and maintaining electrochemical performance in cold environments.

2.5 Enhanced Lithium Plating and Dendrite Formation Risks.

The morphology of lithium deposition varies with temperature, influenced by distinct nucleation and SEI characteristics of lithium metal anodes. For example, SEI formed below -80 °C in DOL/DME electrolytes is thinner and more chemically stable than SEI formed at room temperature. However, the SEI's ionic conductivity decreases at -30 °C, with low-temperature-induced structural changes increasing resistance and limiting Li⁺ diffusion, significantly impacting battery performance.¹⁵

At low temperatures, lithium-ion diffusion is drastically reduced, leading to lithium plating on the anode surface due to limited ion availability and reaction kinetics. This phenomenon risks forming dendrites that can cause internal short circuits, posing severe safety hazards. Designing electrolytes with antiplating additives and ion-dispersing properties is essential for mitigating lithium dendrite formation and ensuring the stable and safe operation of lithium batteries under low-temperature conditions.

3. Electrolyte Engineering to Improve Low-Temperature Performance of LBs

3.1 Modifying Lithium Salts to Enhance Electrolyte Properties

The performance of electrolyte is significantly influenced by the presence of lithium salts, which affect the dissociation, solubility, and ionic conductivity of the SEI. Additionally, lithium salts can also reduce the freezing point of the electrolyte via solvent colligative effects.

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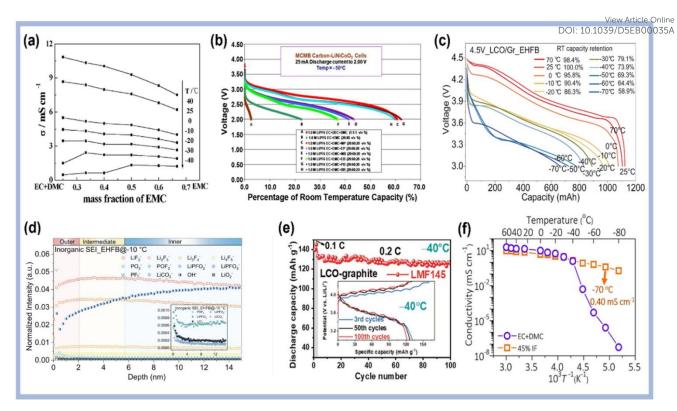


Fig. 4 a) lonic conductivity of 1 M LiPF₆ in EC/DMC/EMC with different EMC ratio at various temperatures. Reproduced from Ref. [25]. Copyright 2004, Elsevier. b) Discharge capacity of Carbon-LiNiCoO₂ cells at -50 °C and C/16 using 1.0 M LiPF₆ in EC/EMC/X (20:60:20 vol.%) (where X= MP, EP MB, EB, PB. and BB). Reproduced from Ref. [28]. Copyright 2010, Institute of Physics Publishing. c) with RT charge-LT discharge protocol. Reproduced from Ref. [30]. Copyright 2023, Springer Nature. 9) Topics of inorganic SEI functional groups after prolonged cycling (EHFB electrolyte, -10 °C). Reproduced from Ref. [30]. Copyright 2023, Springer Nature. e) Cycling stabilities of LCO||graphite full cell in LMF145 at -60 °C and 0.05C. Reproduced from Ref. [34]. Copyright 2023, Wiley-VCH. f) lonic conductivities of different electrolytes at various temperature. Reproduced from Ref. [35]. Copyright 2023, Royal Society of Chemistry.

3.1.1 LiPF₆ Enhances Thermal Stability and Overall Performance.

LiPF₆ demonstrates high dissociation, leading to the production of LiF and the formation of a LiF-rich inorganic SEI. 16,17 It exhibits minimal corrosion towards aluminum and possesses other favorable properties, making it the most widely used salt in nonaqueous electrolytes. Cho et al. 18 reported a low-temperature electrolyte with methyl propionate (MP) as the solvent. This LiPF₆/MP/FEC electrolyte enabled the battery to successfully cycle 100 cycles at a rate of 0.2C at -20°C without capacity attenuation. Figure 3a illustrates the ionic conductivities of three electrolytes at different temperatures, the M9F1 electrolyte exhibiting the highest ionic conductivity over an extensive temperature range. Moreover, the full cell employing this electrolyte is capable of maintaining 60% of the room temperature capacity (RTC) at -40°C. However, it is sensitive to moisture and reacts with H2O to generate HF, which can corrode electrodes and reduce battery capacity, limiting its lowtemperature application. Consequently, researchers have synthesized various lithium salts to improve low-temperature performance.

3.1.2 Borates (LiBF₄, LiBOB, LiDFOB) Reduce Charge Transfer Resistance.

LiBF₄, LiBOB, and LiDFOB have been utilized as alternatives to LiPF₆ in low-temperature electrolytes. LiBF₄-based electrolytes showed lower RCT compared to LiPF₆-based electrolytes and demonstrated superior capacity retention at -30 °C. However,

the poor ionic conductivity and severe side reactions with the lithium anode led to an unstable SEI. LiBOB has excellent filmforming ability and functions effectively below zero, but its high viscosity limits its low-temperature application. LiDFOB integrates the advantageous properties of LiBOB and LiBF₄, rendering it more appropriate for applications in lowtemperature electrolytes (Fig. 3b). 19 Han et al. 20 proposed a new electrolyte (2.4M LiDFOB/EA/FEC) that solved Li⁺ transport dynamics and stabilized the interface at low temperatures. The LiNi_{0.9}Co_{0.05}Mn_{0.05}O₂ (NCM90)||Li full cell exhibited discharge capacities of 173 mAh g-1 at -40 °C and 152 mAh g-1 at -60 °C with this electrolyte. Additionally, LiDFOB is commonly utilized in binary or ternary salt systems to decrease interface resistance, expand the variety of anions, and improve lowtemperature performance (Fig. 3c). Liang et al.21 investigated the behavior of ternary BF₄, PF₆, and difluoro(oxalato)borate tetrahydrofuran (THF), showing enhanced electrochemical stability and battery performance. The formulated electrolyte achieved 93.4% capacity retention after 100 cycles at -30 °C and 0.05C.

3.1.3. Sulfonylimides (LiFSI, LiTFSI) Improve Ionic Conductivity.

LiFSI possesses high solubility, excellent conductivity, and fast desolvation kinetics, which enables its extensive application in low-temperature electrolytes. Jin et al.²² designed an electrolyte containing 1M LiFSI in DOL/DME, noting that low-affinity solvents facilitate fast desolvation. The Li||NCM523 cells with this electrolyte retained 66% capacity at -40 °C

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compared to room temperature. However, LiFSI poses environmental risks and high preparation costs. LiTFSI-based electrolytes, with higher ionic conductivity and improved interfacial stability, have garnered attention. Ma et al. reported ether-based electrolyte containing 1M LiTFSI dimethoxymethane (DMM), which showed better coordination between Li-ions and anions, lower desolvation energy, and favorable SEI formation (Fig. 3d). The Li||SPAN full cell demonstrated stable cycling for 500 cycles and maintained an average CE of 97.67% at -20 °C. At -40 °C, the cell exhibited an initial discharge capacity of 422.3 mAh g⁻¹ and retained 63.8% capacity after 120 cycles at 0.1C.23 Nevertheless, the anion of LiTFSI can corrode aluminum current collectors under high voltage, compromising the cycling stability and safety of batteries. Additionally, the mechanism by which TFSI⁻ forms the CEI remains unclear. These limitations underscore the need to develop novel sulfonimide salts for advanced electrolyte systems.

3.1.4. Asymmetric Salt (LiClO₄) Lowers the Freezing Point.

The structure of lithium salts also impacts electrolyte performance. Wang et al. investigated the influence of asymmetric lithium salts (lithium perchlorate (LiClO₄)) in an ES/10% FEC electrolyte, and discovered that augmenting the asymmetric structure would reduce the freezing point.²⁴ The ES/FEC electrolyte remained liquid at -40 °C. Using this optimized electrolyte, the NCM811||Li full pouch cell demonstrated 83.3% of RTC at -20 °C (Fig. 3e).

3.2. Modifying the Solvent to Improve Li*-Solvent Interactions.

The solvent is a critical component of low-temperature electrolytes, and its design aims to reduce viscosity and freezing point, enhance ionic conductivity, and modify the solvation structure to accelerate the kinetics of desolvation. Solvents can be categorized as follows:

3.2.1 Conventional Solvents.

Carbonate-Based Solvents for Lowering Freezing Points. Carbonate-based solvents are frequently combined with other solvent types to form binary co-solvent systems, which exhibit significantly lower melting points compared to their individual components. Ai et al. 25 found that co-solvents with low melting points effectively broaden the operational temperature range of the electrolyte (Fig. 4a). Carbonates can also form quaternary multicomponent systems when combined with other solvents.^{26,27} Smart et al.²⁸ investigated diverse esters, such as methyl propionate (MP), ethyl propionate (EP), and methyl butyrate (MB), as constituents of the electrolyte (Fig. 4b). The electrolyte containing MP exhibited exhibits excellent capacity at 0.1C and -50°/-60°C. Another electrolyte composed of ethyl acetate (EA) exhibits an ionic conductivity of 1.7 mS cm⁻¹ at -40°C and enables the full cell to retain 90% of RTC.29 Chen et al.30 investigated the effects of ethyl acetate-based solvents with different degrees of fluorination and fluorine substitution sites. They measured the discharge capacity of the battery at different temperatures and analyzed the composition of the SEI after long-term cycling under low-temperature conditions using

Ester-Based Solvents to Facilitate Ion Migration. Compared to carbonates, low-molecular-weight esters exhibit lower melting points, viscosities, and higher dielectric constants. These properties enable esters to serve as efficient electrolyte solvents, promoting rapid Li⁺ ion transport at low temperatures.31 Initially, esters were used as co-solvents in EC and PC-based electrolytes. Li et al. 32 developed an EA-based electrolyte by combining high-concentration electrolytes (HCE) and additive strategies. The low freezing point of EA stabilizes Li* transport at low temperatures, while the high concentration of LiPF₆ and FEC ensures the formation of a LiF-rich inorganic SEI and suppresses side reactions between EA and Li. Methyl propionate, which has a molecular weight similar to ethyl acetate, also demonstrates excellent low-temperature performance.33

Additionally, the introduction of a diluent to the ester-based electrolyte to form a low-temperature high-concentration electrolyte (LHCE) can further enhance electrolyte performance. Lei et al.³⁴ found the presence of fluorobenzene (FB) promotes anion entry into the solvation shell through dipole-dipole interactions and increases the proportion of stable free solvent molecules. With the diluent, methyl acetate (MA) can act as the sole electrolyte solvent to improve ultra-low temperature performance in LIBs (Fig. 4e). Liu et al. introduced iso-butyl formate (IF) as an anti-freezing agent in DMS-based electrolytes, the ionic conductivity of the electrolyte containing 45% IF shows a slower decline at low temperatures (Fig. 4f).³⁵

Ether-Based Solvents to Reduce Viscosity and Improve Anode Compatibility. Compared with carbonates, ethers possess ultralow freezing points, excellent compatibility with lithium metal, and rapid Li⁺ transport kinetics.³⁶ These advantages have made ethers a preferred choice for low-temperature electrolytes. ethers are susceptible to oxidation decomposition at high potentials, which limits their application range.37,38 Holoubek et al. selected diethyl ether (DEE) as the sole solvent for LIBs (Fig. 5a).³⁹ They proposed that the solvation structure of the electrolyte governs charge-transfer behavior at low temperatures, owing to the weak solvation nature of the solvent. A similar effect was observed with dimethyl ether (DME) as a solvent.40 The addition of dibutyl ether (DBE) significantly enhanced the low-temperature performance of Li-S cells due to its weak Li* solvent-binding interactions, promoting a highly ionpaired solvation structure at low salt concentrations.41 The backbone structure of ethers also influences their oxidation stability. Chen et al. 42 investigated the correlation between the ion-solvent coordination extent in the electrolyte and the electrochemical behavior through comparing ethers featuring distinct -(CH2)_n- chain lengths, such as DME, 1,3dimethoxypropane (DMP). They found that DMP, featuring a five-membered chelating ring, can form a more stable sixmembered chelate complex with Li+, thereby significantly enhancing Li⁺ solvation and effectively reducing side reactions with labile free solvent molecules (Fig. 5b).

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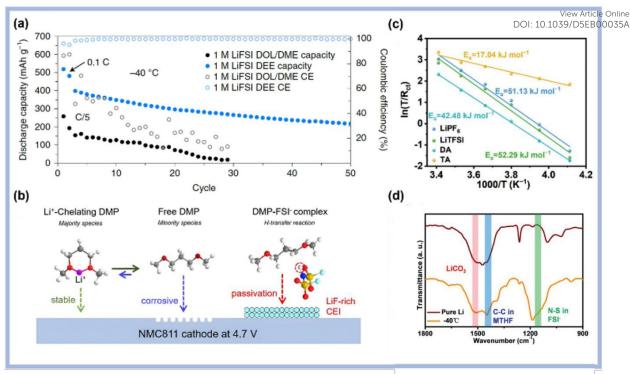


Fig. 5 a) Cycling performance using different electrolyte at -60 °C and 0.2 C. Reproduced from Ref. [39]. Copyright 2021, Springer Nature. b) Electrolyte species and their roles at the cathode surface. Reproduced from Ref. [42]. Copyright 2023, Wiley-VCH. c) The activation energies of Rct fitted by the Arrhenius equation from 20 to -30 °C for different electrolytes. Reproduced from Ref. [44]. Copyright 2024, Wiley-VCH. d) FTIR spectra of Li metal anodes before and after 10 cycles at -40 °C. Reproduced from Ref. [45]. Copyright 2022, Wiley-VCH.

Yoon et al. developed a ternary solvent low-temperature electrolyte based on ether / hydrofluoroether fluoroethylene carbonate (FEC).⁴³ Compared to other ethers, tetrahydrofuran (THF) has a more symmetric structure, reducing the electron density of the oxygen atom and enhancing solubility for LiNO3, which has garnered significant attention. The electrolyte composed of LiFSI-LiNO₃/THF maintained a high conductivity of 2.87 mS cm⁻¹ at -60 °C. Liang et al.44 proposed a THF-based ternary-anion electrolyte to improve electrolyte kinetics at low temperatures. They found that the interaction between the three anions was weakened by repulsive forces, leading to predominant coordination of Li*-NO3- in the Li⁺ solvation structure (Fig. 5c). At -60°C, the optimized electrolyte exhibited high ionic conductivity of 3.39 mS cm⁻¹. In a binary ether-based electrolyte (1 M LiFSI in MTHF/THF, containing 1 wt% LiNO₃), the Li||CoSeOx cell demonstrated superior low-temperature performance.⁴⁵ The weak Li*-solvating MTHF reduced the kinetic barrier for Li* desolvation, while THF, with a high donor number, enhanced the solubility of LiNO₃, resulting in high ionic conductivity while maintaining weak Li*-solvation effects. FTIR analysis revealed the presence of solvent-derived C-C bonds and FSI-derived N-S bonds within the SEI (Fig. 5d).

Naoi et al.⁴⁶ proposed a non-flammable hydrofluoroether as a solvent for low-temperature applications. The branched hydrofluoroether 2-trifluoromethyl-3-methoxyperfluoropentane (TMMP) can be incorporated into the carbonate-based electrolyte, reducing the melting point of the electrolyte and concurrently generating a low-surface-energy SEI to enhance the transport rate of lithium ions at low

temperatures. At -20 °C, the electrolyte containing TMMP enables the capacity of the MCMB||LCO battery to remain 60% of RTC, approximately 20% higher than the base electrolyte. However, ether-based solvents suffer from poor oxidative stability and tend to decompose under high-voltage conditions, resulting in irreversible capacity loss and degraded cycling performance. These limitations necessitate the development of multi-component solvent systems or novel electrolyte formulations to address these challenges.

Nitrile-Based Solvents to Lower Solvation Energy. Researchers have also explored organic solvents containing sulfur (S), nitrogen (N), or phosphorus (P) for low-temperature electrolytes. A7,48 Nitrogen-containing solvents, in particular, have garnered attention for their ability to reduce interfacial impedance and facilitate the formation of a highly efficient solid electrolyte interphase (SEI) layer, making Nitrides a promising candidate for low-temperature applications.

In contrast to carboxylic acid esters and ether solvents, nitriles exhibit higher molecular polarity and dielectric constants, which makes them a distinctive type of electrolyte co-solvent for LBs. Luo et al.⁵⁰ prepared an electrolyte with isobutyronitrile (iBN) as a co-solvent. The resulting electrolyte, consisting of 8.33 vol% ethylene carbonate (EC), 31.67 vol% ethyl methyl carbonate (EMC), and 60 vol% iBN, demonstrated a sufficiently high conductivity of 1.152 mS cm⁻¹ at -70 °C. The adiponitrile (ADN)-based electrolyte consisting of LiFSI and LiODFB in ADN/EC, could expand the operational temperature window.⁵¹ Lu et al. found the use of fluoroacetonitrile (FAN)

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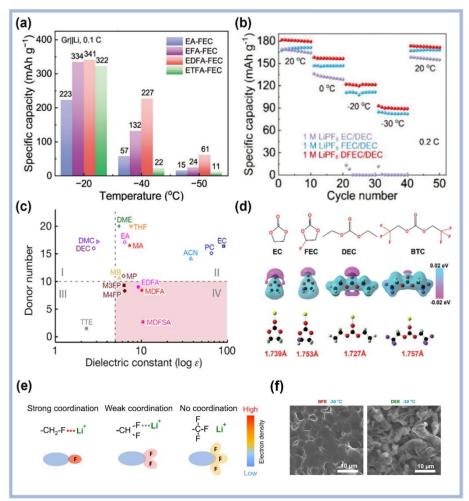


Fig. 6 a) The discharge capabilities of pouch cells with EDFA-FEC and EA-FEC at 0.2 C under different temperatures after fully charging under 25 °C. Reproduced from Ref. [59]. Copyright 2023, Wiley-VCH. b) Rate performance of NCM811||Li cells at 0.2 C from 20 to -30 °C. Reproduced from Ref. [60]. Copyright 2021, Wiley-VCH. c) The solvent diagram of DN versus dielectric constant. Reproduced from Ref. [61]. Copyright 2023, Springer Nature. d) Molecular structure, charge distribution and optimised structure of liquid solvents. Reproduced from Ref. [62]. Copyright 2023, Elsevier. e) The Coordination chemistry of monofluoride, diffuoro, and trifluoro groups. Reproduced from Ref. [64]. Copyright 2023, Springer Nature. f) SEM images of plated Li on Cu foil in BFE and DEE electrolytes at -30 °C, respectively. Reproduced from Ref. [64]. Copyright 2023, Springer Nature. from Ref. [64]. Copyright 2023, Springer Nature.

enabled the formation of a small solvation sheath, which facilitated fast solvation and desolvation processes. 52 The FANbased electrolyte exhibited high ionic conductivity of 11.9 mS cm⁻¹ at -70 °C. While nitrile-based electrolytes exhibit excellent low-temperature performance, they still face critical challenges such as poor reductive stability, severe interfacial side reactions, high toxicity, elevated production costs, and significant environmental risks, all of which demand urgent resolution.

3.2.2 Fluorinated Solvents.

The addition of fluorinated solvents to electrolytes is a widely used strategy to enhance battery performance. Fluorination can reduce the electron density around the oxygen atom in the facilitating the desolvation process electrochemical reactions. It can also decrease the surface tension of solvent molecules, improving electrolyte wettability and reducing contact resistance. Additionally, incorporating fluorinated solvents can enhance interfacial chemistries, promoting faster ion transport at lower temperatures. 53,54

Fluoride Ester-Based Solvents to Improve Interface Compatibility and Enhance Desolvation Rate. Research has shown that fluorinated carboxylic acid esters can enhance the

low-temperature performance of electrolytes.55 The addition of methyl difluoroacetate (MDFA) and ethoxy-pentafluorothe cyclotriphosphazene (PFPN) optimizes coordination environment, thereby accelerating desolvation kinetics. Yang et al. examined the evolution of the SEI at various temperatures using an electrolyte containing the weakly solvated molecule ethyl trifluoroacetate (ETFA).56 Electrolytes containing MDFA, PFPN, and fluoroethylene carbonate (FEC) maintain high capacity at -50°C, while those containing ETFA achieve high reversible capacity at -30°C. Furthermore, LiFSI electrolytes in pure ETFA exhibit a wide electrochemical window and low solvation energy.

Fluorinated esters can also be incorporated as co-solvents into traditional carbonate electrolytes to enhance lithium salt solubility, regulate solvation structure, and stabilize the interface. 57 Cui et al. 58 introduced monofluorinated ester (MTFA) into a LiPF₆-dimethyl carbonate-fluoroethylene carbonate system. The resulting electrolyte enabled a 1 Ah cell to maintain 0.65 Ah at a 2 C discharge rate and showed virtually no capacity fade at a 0.2 C after 80 cycles at -20 °C. Mo et al. 59 explored the influence of fluorination degree on electrolyte performance, they found that ethyl difluoroacetate (EDFA), with a moderate

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degree of fluorination, exhibited better salt dissociation and a more balanced solvation structure compared to highly fluorinated solvents. The EDFA-FEC electrolyte facilitate rapid desolvation, high ionic conductivity, and promote the fast diffusion of Li⁺ in the SEI, providing excellent low-temperature performance (Fig. 6a). Wang et al.⁶⁰ investigated the influence of different degrees of fluorination in ethylene carbonate derivatives on the Li⁺ solvation shell. They found that the degree of fluorination affected ion-dipole interactions, with DFEC demonstrating faster ion desolvation behavior compared to non-fluorinated solvents like EC (Fig. 6b). Xu et al.⁶¹ designed a 1M LiFSI electrolyte in a mixture of MDFA and MDFSA, which enabled full cell to retain 93.9% of its capacity after 260 cycles at -30 °C and 0.1C, with an average Coulombic efficiency (CE) of 99.98% (Fig. 6c). Xiao et al.⁶² developed a fluorinated carbonate electrolyte using bis(2,2,2-trifluoroethyl) carbonate (BTC) as the solvent (Fig 6d). The fluorination of the solvents improved anodic stability at high voltages and accelerated Li+ desolvation, resulting in excellent electrochemical kinetics.

Fluoride Ether-Based Solvents to Regulate Solvation Structure and Broaden the Electrochemical Window Fludpinated ભામિક્ક due to their enhanced antioxidation properties and improved interface compatibility, have been explored for electrolyte applications. 63,64 Zhang et al. 65 synthesized monofluorinated bis(2-fluoroethyl) ethers (BFE) as solvents to dissolve LiFSI. The resulting electrolyte possesses high ionic conductivity and a wide electrochemical window, and enhances the cycling stability of the battery. The monofluoro substitution maximized ion conductivity by optimizing the coordination interaction between fluorine atoms and Li⁺ cations (Fig. 6e). The Li||NCM811 coin cell with this BFE electrolyte showed excellent cycling stability, retaining up to 80% capacity after 300 cycles at 3.5 and 7 mA cm⁻² at 25 °C. Even at -30 °C, the cell maintained over 90% capacity after 150 cycles at 1.75mA cm⁻². As shown in Figure 6f, the BFE electrolyte exhibits more uniform and dense lithium deposition. A mono-fluorinated linear ether, bis(2fluoroethoxy) methane (BFME), with enhanced solvating ability.66 The -OCH2O- structure and fluorine substitution on the

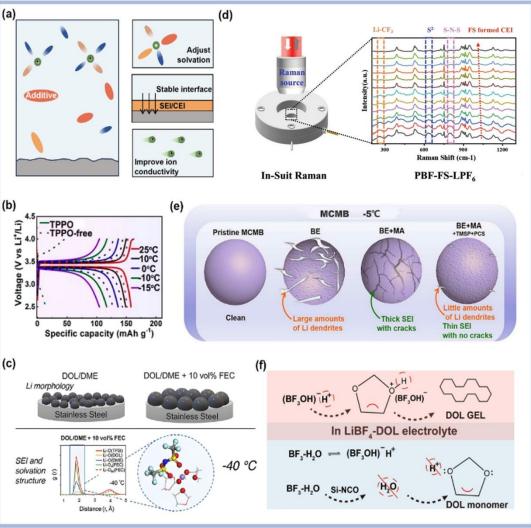


Fig. 7 a) Effects of additives on improving the performance of cell at low temperature. b) the charge–discharge curves using different electrolytes. Reproduced from Ref. [71]. Copyright 2021, American Chemical Society. c) Schematic illustration of the effects of FEC additive on interfaces of cell and the solvation structure of electrolyte at low temperature. Reproduced from Ref. [79]. Copyright 2020, American Chemical Society. d) In situ Raman installation diagram and spectra between LFP cathode and PBF-FS-LPF₆. Reproduced from Ref. [80]. Copyright 2022, Wiley-VCH. e) Schematic illustration of the evaluation of SEI on MCMB in various electrolytes at –5 °C. Reproduced from Ref. [86]. Copyright 2019, Elsevier. f) Schematic representation of the mechanism of water scavenger to prevent DOL-based electrolyte gelation. Reproduced from Ref. [87]. Copyright 2023, Wiley-VCH.

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β-C position enabled BFME to balance the affinity to Li⁺ with improved ionic conductivity and reduced Li⁺ desolvation energy. Fluoride ethers can also be employed as diluents in LHCEs, which do not participate in the solvation structure of Li⁺.66 Jiang et al. developed a fluorinated ether, bis(2,2,2-trifluoroethyl) ether (BTFE).68 The resulting cells delivered 90 mAh g⁻¹ at 0.1C at -20 °C. Although fluorinated solvents facilitate desolvation processes, they exhibit drawbacks including low ionic conductivity, high viscosity, toxicity, and high costs. Therefore, combining them with other solvents is necessary to tailor electrolyte properties.

3.3 Employing Additives to Regulate the Multiple Interfaces of RLBs.

By adding small amounts of specific additives, the high-voltage stability, film-forming ability, and desolvation energy of the electrolyte can be improved, thus boosting its low-temperature performance. Additives can generally be classified as follows.

3.3.1 N-Containing Additives to Suppress Lithium Dendrite Growth.

Lithium nitrate (LiNO₃), due to its higher redox potential compared to other anions or solvents, is preferentially reduced on the lithium surface, forming a robust SEI enriched with Li₃N and Li₂O inorganic components. Li₃N exhibits a relatively high ionic conductivity, coupled with organic components, forms an SEI with outstanding mechanical properties to endure the volume changes, while facilitating ion transport. 69,70 However, the poor solubility of LiNO₃ in carbonates and esters limits its widespread use. To address this, various methods have been employed to enhance its solubility in electrolytes, thereby improving low-temperature battery performance.

Increasing the Solubility of LiNO₃. Xu et al.⁷¹ introduced a synergistic additive, triglyme (G3)-LiNO₃ (GLN), into a carbonate-based electrolyte to form a SEI with uniform Li-ion flux, enabling dendrite-free and reversible lithium plating. Moreover, the introduction of tris(pyrrolidinophosphine) oxide (TPPO), a high DN solvent, facilitated the dissolution of LiNO₃ in the carbonate electrolyte, further enhancing the lowtemperature performance of the cells. TPPO's high DN allows LiNO₃ to dissolve more easily in the electrolyte, promoting the formation of a Li3N-rich SEI film (Fig. 7b).72 Similarly, Zhang et al.73 used DMSO, a high-donor-number solvent (HDNS), to dissolve LiNO₃ and added PC, a low-donor-number solvent (LDNS), to improve the electrolyte oxidative stability. Jiang et al.74 reported that combining LiNO₃ with vinylene carbonate (VC) in an ether-based electrolyte improved the electrochemical behavior over a wide temperature range. The synergistic effects of LiNO₃ and VC, which decompose on the electrode surface upon entering the electrolyte solvation shell, resulted in the formation of a stable bilayer SEI.75 Qiu et al.76 explored the mechanisms behind concentrated salt electrolytes, using LiFSI-LiNO₃-LiFSI ternary salts in a THF solvent. The combination of LiNO₃ and LiFSI helped form stable Li₂O-LiF-rich SEI layers, while LiFSI stabilized the electrolyte under high concentrations. Additionally, sulfolane (SL) demonstrated similar capabilities in dissolving LiNO₃, thereby enhancing electrochemical performance.⁷⁷

Other Additives Beyond LiNO₃. Isosorbide dinitrate (ISDN)_{bin} nitrate compound, can also be used as a Substitute of PinO₃. Zhang et al.⁷⁸ blended ISDN into 1 M LiFSI in DME/HFE, improving the cycle life of high-voltage batteries. The decomposition products of ISDN, including LiN_xO_y, are identified as the key components for enhancing the uniformity of the SEI. In the bilayer SEI, LiN_xO_y generated by ISDN dominates the top layer near the electrolyte, while LiF forms the bottom layer adjacent to the anode. The bilayer SEI improved the uniformity of Li deposition, reducing side reactions of active Li and electrolyte. Furthermore, Zhao et al.⁷⁸ incorporated Cu(NO₃)₂ into an ester-based electrolyte to regulate solvation behavior. The addition of Cu(NO₃)₂ suppressed the transition metals dissolution, voltage decay, and improved the thermal stability of the cathode material.

3.3.2 F-Containing Additives to Govern the Transport of Li⁺ Ions.

Fluoroethylene carbonate (FEC) is renowned for its film-forming capabilities and its ability to reduce on the anode surface before conventional carbonates, resulting in a denser solid electrolyte interphase (SEI). This property has led to its widespread use in enhancing interfacial stability. Thenuwara et al.80 investigated the use of FEC-containing electrolytes for low-temperature battery applications (Fig. 7c). Compared to the DOL/DME electrolyte without additives, the addition of 10% FEC resulted in the formation of a stable, highly Li*-conductive SEI. The FECmodified electrolyte enabled cycling at temperatures as low as -60°C with reasonable CE, which declined from 85% to 50% over 50 cycles. Notably, the SEI formed with the FEC-containing electrolyte was richer in inorganic species (LiF, Li₂CO₃) and thinner than that formed with the baseline electrolyte at both -20°C and -40°C. Additionally, several new fluorine-containing additives have been proposed. For instance, a commercial LiPF₆/EC/DMC electrolyte combined with 4,4'-sulfonyldiphenol (FS) and perfluoro n-butylsulfonyl fluoride (PBF) additives (referred to as PBF-FS-LPF₆) enabled lithium-metal batteries (LMBs) to operate at -40°C. The fluorinated additive PBF altered the solvent sheath structure of the carbonate electrolyte and formed a fluorine-rich (LiF) SEI.81 The emergence of highly electronegative sulfur (S) and FSI-derived S-N-S structures can be observed via in situ Raman spectroscopy (Fig. 7d), which effectively suppresses the decomposition of LiPF6 and thereby stabilizes the CEI.

3.3.3 S-Containing Additives to Facilitate Uniform Li Deposition.

Sulfur-containing additives can also function as film-forming agents, enhancing the stability and conductivity of the SEI. 82 Lan et al. demonstrated that the incorporation of N-tert-butyl-2-thiophenesulfonamide (NTSA) as a multifunctional electrolyte additive improved the electrochemical performance of $\text{LiCoO}_2 \mid \mid \omega - \text{Li}_3 \text{V}_2 \text{O}_5$ full cells across a wide temperature range, primarily due to enhanced interfacial stability. 82 This performance enhancement was attributed to the decomposition of NTSA on the electrode surface, resulting in the formation of a uniform and robust electrode/electrolyte interphase. This interphase was enriched with multiple inorganic compounds, including LiF, Li_3N, and Li_2S, on both the

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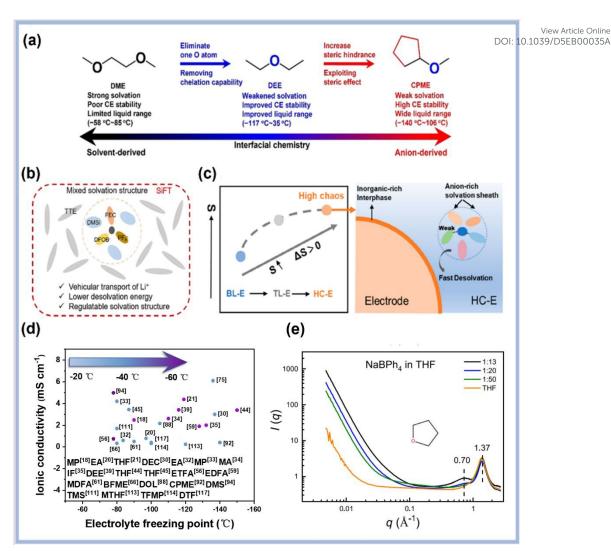


Fig. 8 a) the solvation structure of SiFT electrolyte. Reproduced from Ref. [91]. Copyright 2023, Elsevier. b) Optimization mechanism of the High chaos electrolyte. Reproduced from Ref. [94]. Copyright 2023, American Chemical Society. c) The binding energies between Li* and solvents/anions obtained by first-principles calculations. Reproduced from Ref. [95]. Copyright 2023, Wiley-VCH. d) Summary of ionic conductivity at low temperature and freezing points of popular electrolyte. e) SAXS data of NaBPh₄ dissolved in THF with different salt concentrations. Reproduced from Ref. [126]. Copyright 2023, Springer Nature.

cathode and anode surfaces. The inorganic-rich interface not only reduced the impedance of Li⁺ migration at low temperatures but also improved the thermal stability of the interface, thereby enhancing the overall stability of the battery at both room and low temperatures. Another sulfur-containing additive, dimethyl sulfide (DMS), has also been employed to improve the low-temperature performance of electrolytes.84

3.3.4 P-Containing Additives to Improve Battery Thermal Stability.

Phosphorus-containing additives are often recognized for their superior flame-retardant properties and can be employed to prevent thermal runaway caused by lithium plating or lowtemperature preheating. Liao et al.85 investigated the effect of difluorobis(oxalato) phosphate (LiDFBOP) electrochemical behavior at low temperatures. Their results showed that a graphite||NCM523 pouch cell with 1 wt% LiDFBOP maintained 93% of its initial capacity after 50 cycles at -20 °C under 0.5 C, significantly outperforming the baseline electrolyte.85 This performance was attributed to LiDFBOP's

preferential oxidation on the cathode and reduction on the anode, forming a thin SEI rich in LiF, Li₂C₂O₄, and LixPOyFz, which ensured SEI stability and fast ionic conduction. Similarly, the addition of 1 wt% LiPO₂F₂ in the electrolyte resulted in 71.9% capacity retention at -20 °C in a graphite||NCM523 cell, compared to only 49.4% for cells with the baseline electrolyte. Functional additives can also inhibit the high reactivity of esters with the anode. Xu et al.87 explored the influences of (trimethylsilyl) phosphite (TMSP) and 1,3-propanediol cyclic sulfate (PCS) as additives in LiPF₆-based electrolytes. This electrolyte expanded the electrochemical window of lithiumion batteries (LIBs) to 3.5-5V and enabled the batteries to function within a wide temperature range from -60°C to 50°C. Due to the lower lowest unoccupied molecular orbital (LUMO) of TMSP and PCS, they decompose prior to methyl acetate (MA) and carbonate solvents, generating an interface rich in P-O and ROSO₂Li, which suppresses the reaction of solvents and mesocarbon microbeads (MCMB). This facilitated the formation of a high-ionic-conductivity and stable SEI on the MCMB anode surface.

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3.3.5 Other Additives.

Jiang et al.⁸⁸ proposed using an electrophile, trimethylsilyl isocyanate (SiNCO), as a water scavenger to inhibit the side reactions of 1,3-dioxolane (DOL), thus enabling DOL to remain liquid within a broad temperature range. SiNCO eliminates moisture by inhibiting the proton-induced ring-opening polymerization of the DOL electrolyte through a nucleophilic addition reaction. Analysis of the CEI revealed that the SiNCO additive contributed to the formation of a thin and inorganic-rich CEI.

Ιi al. investigated the effect of et hexafluorophosphate (CsPF₆) as a film-forming additive in lowtemperature electrolytes.[88] Cs+ can enrich and promote the decomposition of ethylene carbonate (EC), forming a highly protective SEI, which in turn inhibits the decomposition of the solvent propylene carbonate (PC) and accelerates the migration of Li*. Additionally, allyl sulfide (AS) has been identified as an electrolyte additive to reduce charge transfer resistance. [90] The incorporation of AS improved the discharge capacity of graphite electrodes, achieving capacities three times larger than those of cells with electrolyte without AS at -30 °C.

3.4 Regulating the Relationships Among Li*, Solvent, and Anions to Manipulate Solvation Chemistry.

Previous research has shown that the reduction in battery capacity and the increase in interface impedance at low temperatures are primarily due to sluggish desolvation, which constitutes the main kinetic barrier at the interface. Consequently, minimizing the desolvation energy barrier and optimizing the electrolyte solvent chemistry are key strategies for improving low-temperature battery performance.

3.4.1 Reduce Solvent Polarity to Regulate the Binding Energy of Li * -Solvent.

Weakly Solvated Electrolytes (WSE). Recently, weakly solvated electrolytes (WSE) have emerged as a promising alternative for achieving solvated structures and properties similar to those of conventional systems, without the constraints of Li salt or diluent. WSEs contain solvents with weak solvating power, which are unable to fully dissociate lithium salts, leading to partial separation of cations and anions. This results in weaker interactions with Li+ and allows more anions to coordinate with Li⁺, forming abundant contact ion pairs (CIPs) and aggregates (AGGs).91 Due to the strong coordination of Li⁺ ions by anions rather than solvent molecules, the preferential formation of anion-derived inorganic SEI or CEI layers occurs when solvated Li* clusters are oxidized or reduced at the electrode/electrolyte interfaces. Examples of WSE: Cyclopentylmethyl ether (CPME): A non-fluorinated ether solvent designed by Zhang et al., which exhibits weak solvating power and remains liquid over a wide temperature range, enabling the tuning of solvating power and physicochemical properties (Fig. 8a).92 2) Bis(2-methoxyethoxy)methane (BME): Developed by Wu et al., this non-fluorinated solvent uses a bi/tridentate coordination strategy to regulate solvation structures. BME has multiple oxygen sites that provide bi/tridentate chelation with Li⁺ ions, forming an anion-rich Li* solvation shell.⁹³ Although WSEs exhibit lower desolvation energy, they suffer from reduced ionic conductivity and 300mpromised reductive stability. Thus, incorporating co-solvents or additives becomes essential to balance ionic transport efficiency and interfacial stability.

Strongly Solvated Electrolytes with Dual Lithium Salts. Zhao et al. proposed a strongly solvated electrolyte in dimethyl sulfite (DMS), which balances fast Li⁺ conduction and efficient desolvation, resulting in a unique solvation structure. ⁹⁴ LiFSI is highly dissociative in DMS, ensuring rapid Li⁺ conduction. Additionally, the high affinity between difluoride (oxalate)borate anions (DFOB-) and Li⁺ accelerates desolvation over a wide temperature range.

Limitations of WSEs. To address issues such as narrow electrochemical windows and poor ionic conductivity in electrolytes composed of weak solvents, the "strong-weak mixed solvent (HSWSS)" strategy has been proposed. This approach designs multicomponent, hierarchically solvating electrolytes (HSE), such as the LiPF₆-LiDFOB-DMSI-FEC-TTE electrolyte, which expands the electrochemical window, improves ionic conductivity, and enhances dendrite suppression (Fig. 8b).⁹⁴

Highly Entropic Electrolytes.

Chen et al.⁹⁶ developed a highly entropic electrolyte rich in multiple anionic solvation structures that weakens Li⁺ solvation, accelerates Li⁺ desolvation at low temperatures, and improves charge transfer kinetics while inhibiting lithium dendrite growth (Fig. 8c). The electrolyte exhibits a self-adapting double-layer solvation structure, where free solvents weaken the binding energy between Li⁺-solvent and Li⁺-anion, while maintaining fast desolvation kinetics and low-temperature adaptability .⁹⁷ Using PTE312, full cells demonstrated outstanding capacities of 79% and 62.5% at -40 °C and -50 °C, respectively.

3.4.2 Low-Concentration Electrolyte (LCE) to Regulate Lithium Deposition Behavior.

Low-concentration electrolytes with low viscosity promote the formation of an anion-derived SEI and facilitate uniform lithium deposition on the lithium anode. 99-100 Wang et al. introduced an innovative ultralow-concentration electrolyte (ULCE). At -60 °C, the LMB with ULCE still exhibited a capacity of approximately 115 mAh g⁻¹, retaining about 57% of RTC. 101 However, low lithium salt concentration in electrolytes reduces ionic conductivity, while excessive solvent content compromises stability and increases flammability. Addressing these challenges necessitates the use of non-solvating co-solvents or novel lithium salts to mitigate these limitations.

3.4.3 Highly Concentrated Electrolytes (HCEs) to Form Anion-Derived SEI.

At high salt concentrations (typically 3-5 M), Li* ions coordinate with all solvent molecules and anions, creating a unique solution structure with negligible free solvent molecules. This structure contrasts sharply with conventional dilute solutions, which are dominated by free solvent molecules. In highly concentrated electrolytes (HCEs), the lowest unoccupied molecular orbital (LUMO) of the Li*-solvation structure shifts from solvents to anions, resulting in an anion-derived SEI. Wang

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 ${\sf Table \ 1 \ A \ summary \ of \ recently \ reported \ low-temperature \ electrolyte \ for \ RLBs.}$

Electrolyte formulation	Cathode/anode	Working temperature	Battery performance	Ref.
Salt engineering				
1M LiPF ₆ in MP:FEC (9:1,vol)	NCM111 graphite 2.7-4.3	-40 to RT	-20 °C @ 0.1C 92% after 100 cycles	18
2.4 M LIDFOB in EA/FEC	NCM9055 Li 2.8-4.3	-60 to RT	-40 °C @ 0.1C 100% after 100 cycles	20
$\label{eq:Lipfob} {\sf LiPF_6: LiBF_4: LiDFOB (2:2:6, molar)} \\ {\sf in THF} \\$	NCM811 Li 2.7-4.3	-30 to RT	-30 °C @ 0.1C 93.4% after 100cycles	21
1 M LiFSI in DMM	SPAN Li 1.8-2.3	-40 to RT	-40 °C @ 0.1C 99.7% after 120 cycles	23
1M LiClO ₄ in ES /10%FEC	NCM811 Li 2.7-4.3	-50 to 70	-33 °C @ 0.1C 80% after 100 cycles	24
Solvent engineering				
Conventional Solvents				
1M LiPF $_6$ in EC:PC:DEC:EMC:EHFB (2:1:1.5:4:1.5 ,vol)	LCO graphite 3.0-4.5	-100 to RT	-10 °C @ 0.1C 98% after 200 cycles	30
3 M LiPF ₆ in EA:FEC (9:1,vol)	NLO graphite 2.7-4.3	-40 to 60	-20 °C @ 0.2C 100% after 1400 cycles	32
1 M LiFSI in MP:FEC (9:1, vol)	LFP graphite 2.5-4.2	-80 to 80	-40 °C @ 0.1C 67.7% after 100 cycles	33
1M LiFSI in MA: FB (4:5,vol)	LCO graphite 3.0-4.3	-60 to RT	-60 °C @ 0.05C 88.5% after 50 cycles	34
1M LiDFOB in DMS/IF/FEC (35:45:20,vol)	LCO Li 3.0-4.4	-70 to 60	-70 °C @ 1/15 C 99% after 170 cycles	35
1M LiFSI in DEE	SPAN Li 1.8-2.3	-60 to RT	-40 °C @ 0.2C 60% after 50 cycles	39
1 M LiFSI in DOL/TTE 10 wt%FEC	NCM622 Li 3-4.3	-40 to RT	-40 °C @ 0.05C 76.1% after 15 cycles	43
$0.5~\mathrm{M}$ LiPF $_6$ $0.5~\mathrm{M}$ LiFSI $0.1~\mathrm{M}$ LiNO 3 in THF/FEC(9:1 vol)	NCM811 Li 2.7-4.3	-60 to RT	-30 °C @ 0.05C 78.6% after 50 cycles	44
1 M LiFSI in MTHF/THF (1:6, vol) 1 wt% LiNO ₃	CoSeOx Li 1.4-3.5	-40 to RT	-40 °C @ 400 mAg ⁻¹ 84% after 100 cycles	45
Fluorinated Solvents				
1.3 M LiFSI in FAN	Graphite Li 2.8-4.5	-80 to RT	-60 °C @ 0.2C 80% after 350 cycles	52
1 M LiPF ₆ in MDFA/PFPN/FEC	NCM811 graphite 2.7-4.4	-50 to 60	-20°C @ 0.2C 87.4 % after 100 cycles	55
1 M LiFSI in ETFA	LTO Li 1.8-2.8	-70 to RT	-40 °C @ 0.05C 99% after 50 cycles	57
1 M LiFSI in EDFA/FEC (9:1, vol)	NCM811 graphite 2.7-4.5	-40 to RT	-20 °C @ 0.2C 93% after 30 cycles	59

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Table 1 (Contd.)			DOI: 10.1	View Article C
Electrolyte formulation	Cathode/anode	Working temperature	Battery performance	Ref.
1 M LiPF $_6$ in DFEC/DEC (1:1, vol)	NCM811 Li 2.7-4.3	-30 to RT	-30 °C @ 0.2C 100% after 50 cycles	60
1 M LiFSI in MDFA/MDFSA/TTE	NCM811 graphite 2.7-4.5	-60 to 60	-30 °C @ 0.2C 82.8% after 350 cycles	61
1 M LiFSI in BFE	NCM811 Li 2.8-4.4	-60 to 60	-30 °C @ 1.75 mA cm ⁻² 90% after 150 cycles	65
1 M LiFSI in BFME	LFP Graphite 2.7-4.3	-60 to 60	-20 °C @ 0.2C 92.5% after 200 cycles	66
Additive engineering				
1M LiPF ₆ in EC/EMC+FEC+LiNO ₃ in TPPO	LFP Li 2.5-4.0	-15 to 70	-15 °C @ 0.1C 100% after 100 cycles	72
1.6 M LiFSI in THF/MTHF (1:1 vol) with 2 wt% LiNO $_3$	LFP Li 2.5-4.0	-40 to RT	-30 °C @ 0.1C 100% after 200 cycles	75
1M LiPF $_6$ in EC/DMC (1:1 vol + 0.2% LiNO $_3$) + FS and PBF	LFP Li 2.5-4.0	-40 to RT	-40 °C @ 0.5C 90% after 100 cycles	81
1M LiPF $_6$ in EMC:DMC:FEC (4:4:2 vol) + 1%NTSA	LiCoO ₂ ω-Li ₃ V ₂ O ₅ 1.5-4.1	-30 to 80	-20 °C @ $0.2 \ A \ g^{-1}$ 100% after 100 cycles	83
2.5 M LiBF ₄ in DOL/DME(7:3 vol) +Si-NCO	LCO Li 3.0-4.2	-40 to RT	-40 °C @ 0.33C 80% after 150 cycles	88
1 M LiPF ₆ in EC/EMC (1:2 vol) + 1% PhMS	NCM523 graphite 2.7-4.1	-10 to RT	-20 °C @ 0.1C 89% after 100 cycles	90
Solvation regulation				
LiFSI :CPME (1:10, molar)	LFP Li 2.5-4.0	-20 to RT	-20 °C @ 0.2C 90% after 400 cycles	92
1 M LiFSI in BME	LFP Li 2.5-4.0	-10 to 60	-10 °C @ 0.2C 86% after 100 cycles	93
0.6M LiFSI 0.4M LiDFOB in DMS	Graphite Li 2.5-4.3	-78 to 60	-20 °C @ 0.1C -80% after 250 cycles	94
0.25 M LITFSI- 0.75 M LIDFOB in TMS/EA/HFE FEC	NCM523 Li 3-4.6	-80 to 40	-40 °C @ 1C 97.5% after 200 cycles	111
1.5 M LIFSI in THF:MTHF/TTE	LFP Li 2.5-4.0	-40 to RT	-40 °C @ 0.05C 100% after 150 cycles	113
1 M LiFSI in TFMP/DME(7:1, vol)	NCM811 Li 2.8-4.2	-40 to RT	-40 °C @ 0.2C 87% after 100 cycles	114
5m LiFSI/EA + DCM (1:4, vol)	PI Li 1.6-3.0	-70 to RT	-70 °C @ 0.2C 75 % after 100 cycles	115
1 M LiFSI in EMC /FEC/DTF (1.5:1.5:7,vol)	NMC811 Li 2.7-4.8	-40 to RT	-40 °C @ 0.2C 93% after 100 cycles	117

et al. developed a concentrated electrolyte that maintains stable charge-discharge cycling from -20 °C to 100 °C. The electrolyte's stable solvation structure and robust SEI contribute to its excellent electrochemical performance across

a wide temperature range. At low temperatures, the highly Li*conductive SEI compensates for the electrolyte's reduced ionic conductivity, enabling stable cycling performance. 102 Although HCEs offer advantages such as high interfacial stability, a broad

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electrochemical window, and good compatibility, the high salt concentration in HCEs leads to challenges like increased viscosity, elevated freezing point, and poor wettability, which require further optimization.

3.4.4 Localized Highly Concentrated Electrolytes (LHCEs) to Balance Li⁺-Anion-Solvent Clusters.

To address the challenges of high-concentration electrolytes (HCEs), researchers have introduced diluents to create localized high-concentration electrolytes (LHCEs). $^{103\text{-}105}$ The polarity mismatch between non-solvating diluents and solvating solvents drives the aggregation of free anions, lithium ions, and solvent molecules within the non-polar diluent. 106 This process maintains anion-participated solvation sheaths without increasing lithium salt concentration. Such a configuration promotes the formation of an inorganic-rich solid electrolyte interphase (SEI) through anion decomposition, thereby improving battery performance. Chen et al. established design principles for LHCEs based on solvent properties. 109 These principles stipulate that the primary solvent should have a donor number (DN) > 10, while the diluent should have a DN \leq 10.

LHCEs can be conceptualized as droplets of high-concentration electrolyte dispersed in an inert diluent, which neither dissolves lithium salt nor reacts with the solvent. The lithium salt concentration in each droplet exceeds 1 M and increases with the diluent proportion. Consequently, the selection of diluents and the regulation of their proportions are critical for LHCEs performance.¹¹⁰ Lin et al.¹¹¹ developed a multifunctional LHCEs comprising a sulfolane (TMS)/ethyl acetate (EA)-based system, diluted with hydrofluoroether (HFE) and supplemented with fluoroethylene carbonate (FEC) additive. A DMC/EC/TTE mixed electrolyte yielded a discharge capacity of 160.7 mAh g⁻¹ at -30 °C.¹¹² Piao et al.¹¹³ added tris(2,2,2-trifluoroethyl) orthoformate (TTE) to a mixed ether-based electrolyte (MixTHF) to decrease the donor oxygen's electronegativity in ether solvents, promoting anion participation in Li⁺ solvation. Researchers have also explored ether fluorides as alternative diluents to fine-tune solvation structures and improve battery performance. Shi et al.¹¹⁴ synthesized a novel amphiphilic solvent, 1,1,2,2-tetrafluoro-3-methoxypropane (TFMP), for use as an LHCEs diluent. TFMP's lithiophilic segment efficiently solvates Li+, facilitating the electrolyte's self-assembly into a unique core-shell solvation structure. Dichloromethane (DCM), another well-known diluent, 115 enabled the LHCEs to exhibit high ionic conductivity (0.6 mS cm⁻¹), low viscosity (0.35 Pa s), and enhanced oxidative stability at -70 °C. Figure 8d summarizes the ionic conductivity at low temperatures and freezing points of representative electrolytes. Fan et al. 116 used tetrafluoro-1-(2,2,2-trifluoroethoxy)ethane (D2) methoxyperfluorobutane (M3) as diluents to modify allfluorinated electrolytes' solvation structure, enhancing cell performance at low temperatures. Electrolytes comprising LiFSI-FEMC/FEC in D2 and LiBETI FEC/DEC-M3 exhibited enhanced ionic conductivity and lower desolvation energy. Cui et al. chose difluoromethane (DTF) as a co-solvent based on molecular electrostatic potential analysis. 117 DTF's moderate minimum electrostatic potential (-21.0 kcal mol⁻¹), balanced, Lite affinity, preserving the anion-rich Solvation/Detructure. Additionally, the localized concentration strategy can also be applied to ionic liquid electrolytes. Liu et al. proposed a locally concentrated ionic liquid electrolyte([LiFSI]₁[EmimFSI]₂([dFBn]₂) named FEdF, 1,2-difluorobenzene (dFBn)), which enables Li||NCA cells to stably cycle 100 times at -20°C under a 0.1C rate, achieving a capacity retention rate of 85.9%. Although LHCEs can form highly stable interfaces with low viscosity and excellent wettability, challenges such as poor diluent compatibility, high cost, and environmental pollution issues still need to be resolved.

3.4.5 Liquefied Gas Electrolyte to Tune the Viscosity at Low Temperatures.

Meng et al. 119 introduced liquefied gas electrolytes (LGEs) as a novel type of wide-temperature-range stable electrolyte (WSE). utilize liquid hydrogen fluoride, hydrofluorocarbons, as a solvent for salt dissolution. This results in an electrolyte with a wide electrochemical window (-3.23 to 2.47 V) over a broad temperature range. Liquid gases like fluoromethane exhibit significantly lower viscosity than conventional liquid electrolytes at ultra-low temperatures, enabling superior low-temperature electrochemical performance. The ionic conductivity of fluoromethane (FM) solvent with 0.1 mol/L LiTFSI reaches 1.1 mS cm⁻¹ at -60°C, significantly surpassing conventional electrolytes. The addition of CO₂ as a film-forming additive to the original electrolyte results in a robust, Li-carbonate-rich SEI on the anode surface. Subsequent studies by Meng et al. incorporated acetonitrile and tetrahydrofuran (THF) as additive solvents, enhancing LiTFSI solubility in the FM-based electrolyte and reducing cell impedance. 120,121 Lin et al. designed a novel liquefied gas electrolyte (LGE) by incorporating liquefied dimethyl ether (Me₂O) into a mixture of 1,1,1,2-tetrafluoroethane (TFE) and pentafluoroethane (PFE).122 This electrolyte maintains stable ionic conductivity (>1 mS cm⁻¹) across a wide temperature range (-78°C to 80°C) and enables Li/NMC622 batteries to achieve stable cycling over 200 cycles at -20°C, with a capacity retention rate exceeding 90%. While these advancements significantly improved the low-temperature performance of lithium-metal batteries (LMBs), the tests were conducted under high-pressure conditions, and its practical application remains to be explored.

4. Conclusion and Perspective

Traditional lithium batteries exhibit temperature-sensitive properties, which make it challenging to meet performance and cycle life requirements for devices operating under extreme temperatures. According to the Arrhenius equation, the electrochemical kinetics in rechargeable lithium batteries (RLBs) deteriorate as the temperature decreases. This slow kinetics leads to thermodynamic issues, such as lithium plating, which ultimately reduces battery performance. During the charging and discharging cycles of RLBs, charge transport, lithium-ion solvation and desolvation, and interfacial stability are all closely tied to the electrolyte. As a result, the electrolyte plays a crucial

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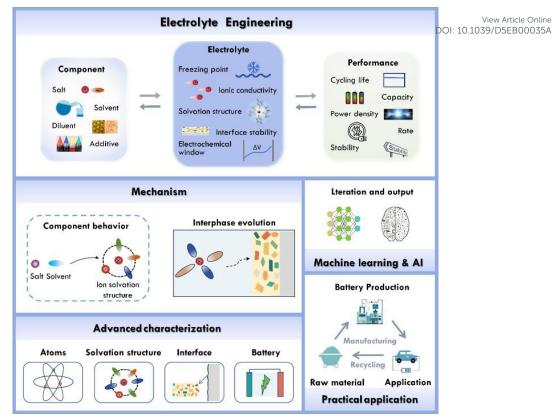


Fig. 9 The research and application opportunities for low-temperature electrolytes. Driven by several factors: (1) deeper insights into underlying mechanisms, (2) Alassisted electrolyte design, (3) advanced characterization techniques, and (4) the need to balance environmental impact with battery performance are discussed.

role in determining the performance of RLBs at low temperatures. Traditional carbonate-based electrolytes exhibit a wide electrochemical window, but their viscosity increases dramatically at low temperatures, rendering them unsuitable for extreme cold conditions (e.g., below -20°C). They are compatible with lithium-ion batteries, cost-effective, and supported by mature manufacturing processes. Ether-based electrolytes demonstrate low viscosity, good compatibility, and high ionic conductivity at low temperatures, functioning even below -40°C. However, their poor oxidative stability limits their use with high-voltage cathodes like NMC811, although they are suitable for lithium-metal batteries. Liquefied gas electrolytes feature ultra-low viscosity and high ionic conductivity over a wide temperature range, making them ideal for broadtemperature lithium-metal batteries. However, they impose stringent sealing requirements on battery systems. In summary, conventional liquid electrolytes require optimized component design to reduce viscosity at low temperatures, enhance ionic conductivity, and broaden the electrochemical window, thereby improving battery performance in cold environments. This review provides a comprehensive summary of the research progress on low-temperature electrolytes for lithium batteries, considering both the individual components and overall systems. Table 1 summarizes representative electrolytes and their battery low-temperature performance. Based on this foundation, we propose novel insights into potential solutions for improving the performance of low-temperature lithium batteries, as illustrated in the figure below (Fig. 9).

4.1 Artificial Intelligence and Machine Learning to Optimize Experiments.

Capturing the intricate electrochemical processes in a battery using simple physical models is particularly challenging, at low temperatures. Furthermore, physicochemical properties of the electrolyte-such as ionic conductivity, Li* migration number, can also impact the battery's performance, including capacity retention rate, cycle life, and energy density, etc. Traditionally, most electrolytes have been designed through a trial-and-error approach, where accumulated experience is a key factor. This method, however, limits the development of new electrolytes and inhibits a deeper understanding of the redox mechanisms and solvation chemistry of electrolytes. Although methods based on dielectric constants and DN values have been proposed for selecting electrolyte components, they remain insufficiently comprehensive. Theoretical frameworks such as DFT and molecular dynamics (MD) have also been explored for electrolyte design, but their models are often too simplistic to accurately predict the specific components needed.

The integration of machine learning (ML) with prior material knowledge known as material knowledge-informed machine learning (MIML) holds significant potential for optimizing battery design. MIML compensates for traditional ML limitations, such as small sample sizes and poor interpretability. 123 For instance, deep learning algorithms have been used to improve the precision of numerical simulations, while neural networks have helped identify correlations

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between atomic polarizability and charge, which facilitates the prediction of molecular dynamics in liquid electrolytes. Additionally, the solvation environment of molecules can be simulated with neural networks, which enhances both the accuracy and speed of the predictive process.

MIML allows the introduction of prior theoretical models to reduce the amount of training data needed, resulting in more accurate outcomes. In recent years, the use of big data

and mechanisms at play, it is essential to observe the behavior of each component at varying spatial scales. Dave et al. developed Clio, a robotic platform that integrates robotics and learning (ML) to facilitate high-throughput machine experimentation and electrolyte property characterization for nonaqueous electrolytes. 124 Guided by experiment-planning algorithms, Clio autonomously optimizes the relationships between ionic conductivity, solvent mass fractions, and salt molality within defined design spaces. For example, Clio successfully optimized conductivity in a ternary solvent system consisting of ethylene carbonate (EC), ethyl methyl carbonate (EMC), and dimethyl carbonate (DMC) with LiPF₆ as the sole salt. This work illustrates the potential of autonomous robotics and integrated testing to accelerate the discovery of electrolytes for energy and materials science applications.

Techniques at the macroscale can offer an overview of the solvation process and identify fault locations, while nanoscale techniques can provide detailed insights into reaction mechanisms. For example, lithium dendrite formation in lithium metal batteries (LMBs) can be studied using optical microscopy, whereas TEM, with its smaller spatial scale, can monitor the progression of SEI or CEI at specific locations. 125 Small-angle Xray scattering (SAXS) can be employed to characterize nanostructures in electrolytes, enabling the analysis of molecular clusters, ion pairs, solvation sheaths cations/anions, aggregates, and domain sizes of the electrolyte (Fig. 8e). Furthermore, when combined with molecular dynamics (MD) simulations, SAXS provides a powerful tool to investigate the working mechanisms of electrolytes. 126 The development of multi-scale characterization methodsaddressing both temporal and spatial dimensions-will likely drive further advances in electrolyte design and improve battery performance.

4.2 Practical Considerations for Electrolytes.

In practical research, it is essential to consider not only the impact of rechargeable low-temperature lithium battery electrolytes on battery performance but also their practical applications. For example, factors such as the influence of electrolyte quantity on energy density in pouch cells, the cost of new electrolytes, production feasibility, and the environmental impact must be taken into account. The potential of new electrolytes is closely linked to the overall battery system. In recent years, commercial sectors such as digital 3C electronics and new energy vehicles have driven the development of LIBs, leading to a reduction in manufacturing and material costs. However, compared to the widely used LiPF₆ electrolyte, the practical application of new electrolytes, such as HCEs (Highconcentration Electrolytes) and LHCEs (Lithium-based High-

concentration Electrolytes), is still constrained by the high cost of lithium salt types and diluents. LiPF2; 1040e39605fts00c35t4 effectiveness and well-established production technology, is the most widely adopted lithium salt in commercial applications, with a current market price ranging from 20 to 30 US dollars per kilogram. However, LiPF₆ exhibits poor thermal stability and is susceptible to hydrolysis, leading to the formation of HF, which adversely affects the lifespan and safety performance of the electrolyte. In contrast, LiFSI, a novel sulfonimide-based lithium salt, demonstrates superior oxidation stability and low interfacial impedance. Nevertheless, constrained by factors such as complex synthesis processes, its cost remains significantly higher at 80 to 100 US dollars per kilogram, approximately 3 to 4 times that of LiPF₆, thus failing to meet large-scale market requirements. To facilitate widespread commercialization, it is imperative to develop continuous synthesis technologies or explore the use of blended lithium salts to effectively reduce costs. Despite their excellent wide-temperature performance and flame-retardant properties, fluorinated solvents may release hydrogen fluoride (HF) or fluorocarbons under high-temperature or abusive conditions, contributing to global warming. These compounds are difficult to degrade via conventional treatment methods. Before commercialization, it is essential to enhance recycling conduct thorough environmental technologies. assessments, and develop low-global-warming-potential (GWP) alternatives to mitigate ecological impacts.

In current research on low-temperature electrolytes, researchers often conduct low-temperature battery testing using small-capacity coin cells, while there are relatively few reports on low-temperature long-cycle testing of high-capacity batteries, such as pouch cells. The stainless-steel casing of coin cells allows them to withstand pressure from gas accumulation during charge-discharge cycles caused by electrode changes and electrolyte consumption. However, due to the material properties of their gaskets, coin cells exhibit higher internal resistance, making them prone to side reactions during cycling, which results in lower capacity retention.

In contrast, pouch cells feature a simpler structure, lower internal resistance, and superior rate performance. Yet, their large surface area and soft aluminum-plastic film outer layer make them susceptible to gas-induced swelling, generating harmful additional stresses that can lead to battery failure. The compact size and limited capacity of coin cells mask deficiencies in electrolyte properties, such as film-forming capability. Furthermore, the electrolyte-to-capacity (E/C) ratios differ significantly between coin and pouch cells, with the latter being substantially lower.127 Therefore, to enhance battery capacity and expand device applications, there is an urgent need for lowtemperature electrolytes with improved film-forming ability and electrochemical stability.

4.3 Non-traditional Systems for Low-Temperature Batteries.

While improvements in electrolytes can enhance the lowtemperature performance of lithium batteries to some extent, many modification strategies have not yet been practical for widespread application. Alternative solutions can be found at the operational level. During battery charging and discharging

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at low temperatures, high resistance leads to the conversion of a significant amount of electrical energy into heat. This heat generated by the charging and discharging process, can be utilized for warming the interior of the battery. This process can be facilitated through Battery Management Systems (BMS) in various ways, allowing the battery to regain performance once its temperature rises above zero degrees Celsius. ¹²⁸

A novel approach that has emerged in recent years is the self-heating battery. In this system, the battery generates heat solely through its internal resistance (i.e., ohmic and polarization resistance). For a single battery, adjusting the discharge current is sufficient to initiate self-heating. Theoretically, self-heating technology can heat the battery faster and more uniformly, offering additional advantages in terms of energy consumption, temperature uniformity, cost, and minimal weight increase.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Data availability

No data was used for the research described in the article.

Notes and references

- 1 Z. Yu, P. E. Rudnicki, Z. Zhang, Z. Huang, H. Celik, S. T. Oyakhire, Y. Chen, X. Kong, S. C. Kim, X. Xiao, H. Wang, Y. Zheng, G. A. Kamat, M. S. Kim, S. F. Bent, J. Qin, Y. Cui and Z. Bao, *Nat. Energy*, 2022, 7, 94–106.
- Z. Li, Y. Yao, S. Sun, C. Jin, N. Yao, C. Yan and Q. Zhang, *Angew. Chem. Int. Ed.*, 2023, **135**, e202303888.
- S. Zhang, Q. Fan, C. Zhang, T. Zhou, K. Kalantar-Zadeh and Z. Guo, Energy Environ. Sci., 2021, 14, 4177–4202.
- 4 E. Fan, L. Li, Z. Wang, J. Lin, Y. Huang, Y. Yao, R. Chen and F. Wu, Chem. Rev., 2020, 120, 7020-7063.
- 5 X. Zhou, Y. Zhou, L. Yu, L. Qi, K.S. Oh, P. Hu, S.Y. Lee and C. Chen, *Chem. Soc. Rev.*, 2024, **53**, 5291-5337.
- 6 D. Hubble, D. E. Brown, Y. Zhao, C. Fang, J. Lau, B. D. McCloskey and G. Liu, *Energy Environ. Sci.*, 2022, **15**, 550.
- 7 N. Zhang, T. Deng, S. Zhang, C. Wang, L. Chen, C. Wang and X. Fan, Adv Mater., 2022, 34, 2107899.
- 8 A. Hu, F. Li, W. Chen, T. Lei, Y. Li, Y. Fan, M. He, F. Wang, M. Zhou, Y. Hu, Y. Yan, B. Chen, J. Zhu, J. Long, X. Wang and J. Xiong, Adv. Energy Mater., 2022, 12, 2202432.

- W. Bai, J. Gao, K. Li, G. Wang, T. Zhou, P. Li, S. Qin, G. Zhang,
 Z. Guo, C. Xiao and Y. Xie, *Angew. Chemy Int.* 159, 12020 1333,
 17647–17651.
- D. Hubble, D. E. Brown, Y. Zhao, C. Fang, J. Lau, B. D. McCloskey and G. Liu, Energy Environ. Sci., 2022, 15, 550-578.
- 11 Y. Tang, Y. Zhang, W. Li, B. Ma and X. Chen, *Chem. Soc. Rev.*, 2015, **44**, 5926–5940.
- 12 Q. Liu and L. Wang, Adv. Energy Mater., 2023, 13, 2301742.
- 13 Y. Sun, B. Liu, L. Liu and X. Yan, Adv. Funct. Mater., 2022, 32, 2109568.
- 14 Z. Piao, R. Gao, Y. Liu, G. Zhou and H. Cheng, Adv. Mater., 2022, 35, 2206009.
- 15 Y. Feng, L. Zhou, H. Ma, Z. Wu, Q. Zhao, H. Li, K. Zhang and J. Chen, *Energy Environ. Sci.*, 2022, **15**, 1711–1759.
- 16 K. Yang, R. Tian, Z. Wang, H. Zhang, Y. Ma, X. Shi, D. Song, L. Zhang and L. Z, Rare Met., 2023, 42, 4128–4141.
- 17 J. Hou, M. Yang, D. Wang and J. Zhang, Adv. Energy Mater., 2020, 10, 1904152.
- 18 Y. Cho, M. Li, J. Holoubek, W. Li, Y. Yin, Y. S. Meng and Z. Chen, *ACS Energy Lett.*, 2021, **6**, 2016–2023.
- 19 S. Zhang, Electrochem. Commun., 2006, 8, 1423-1428.
- 20 R. Han, Z. Wang, D. Huang, F. Zhang, A. Pan, H. Song, Y. Wei, Y. Liu, L. Wang, Y. Liu, J. Xu and X. Wu, *Small*, 2023, **19**, 2300571.
- 21 P. Liang, J. Li, Y. Dong, Z. Wang, G. Ding, K. Liu, L. Xue and F. Cheng, *Angew. Chem. Int. Ed.*, 2024, **03**, e202415853.
- 22 C. Jin, N. Yao, Y. Xiao, J. Xie, Z. Li, X. Chen, B. Li, X. Zhang, J. Huang and Q. Zhang, Adv. Mater., 2022, 35, 2208340.
- 23 T. Ma, Y. Ni, Q. Wang, W. Zhang, S. Jin, S. Zheng, X. Yang, Y. Hou, Z. Tao and J. Chen, *Angew. Chem. Int. Ed.*, 2022, **61**, e202207927.
- 24 Y. Wang, Z. Li, W. Xie, Q. Zhang, Z. Hao, C. Zheng, J. Hou, Y. Lu, Z. Yan, Q. Zhao and J. Chen, *Angew. Chem. Int. Ed.*, 2024, **136**, e202310905.
- L. Xiao, Y. Cao, X. Ai and H. Yang, *Electrochim. Acta*, 2004, 49, 4857–4863.
- 26 H. Li, R. Hua, Y. Xu, D. Ke, C. Yang, Q. Ma, L. Zhang, T. Zhou and C. Zhang, *Chem. Sci.*, 2023, **14**, 10147–10154.
- 27 S. Zhang, Y. Zheng, X. Huang, J. Hong, B. Cao, J. Hao, Q. Fan, T. Zhou and Z. Guo, *Adv. Energy Mater.*, 2019, **9**, 1900081.
- 28 M. C. Smart, B. V. Ratnakumar, K. B. Chin and L. D. Whitcanack, J. Electrochem. Soc., 2010, 157, A1361.
- 29 K. Chen, Z. Yu, S. Deng, Q. Wu, J. Zou and X. Zeng, J. Power Sources, 2015, 278, 411–419.
- 30 Y. Chen, Q. He, Y. Zhao, W. Zhou, P. Xiao, P. Gao, N. Tavajohi, J. Tu, B. Li, X. He, L. Xing, X. Fan and J. Liu, *Nat. Commun.*, 2023, 14, 8326.
- 31 L. Tan, P. Chen, Q. Chen, X. Huang, K. Zou, Y. Nie and L. Li, Rare Met., 2023, 42, 4081–4090.
- 32 Z. Li, N. Yao, L. Yu, Y. Yao, C. Jin, Y. Yang, Y. Xiao, X. Yue, W. Cai, L. Xu, P. Wu, C. Yan and Q. Zhang, *Matter*, 2023, 6, 2274-2292.
- 33 Z. Li, Y. Yao, M. Zheng, S. Sun, Y. Yang, Y. Xiao, L. Xu, C. Jin, X. Yue, T. Song, P. Wu, C. Yan and Q. Zhang, *Angew. Chem. Int. Ed.*, 2024, **137**, e202409409.
- 34 S. Lei, Z. Zeng, H. Yan, M. Qin, M. Liu, Y. Wu, H. Zhang, S. Cheng and J. Xie, Adv. Funct. Mater., 2023, 33, 2301028.
- 35 J. Liu, B. Yuan, N. He, L. Dong, D. Chen, S. Zhong, Y. Ji, J. Han, C. Yang, Y. Liu and W. He, *Energy Environ. Sci.*, 2023, **16**, 1024-1034.
- 36 X. Dai, K. Zou, W. Jing, P. Xu, J. Sun, S. Guo, Q. Tan, Y. Liu, T. Zhou and Y. Chen, *J. Mater. Chem. A*, 2022, **10**, 16152–16162.
- 37 K. Yang, R. Tian, Z. Wang, H. Zhang, Y. Ma, X. Shi, D. Song, L. Zhang, and L. Zhu, *Rare Met.*, 2023, **42**, 4128–4141.
- 38 Q. Fan, J. Jiang, S. Zhang, T. Zhou, W. K. Pang, Q. Gu, H. Liu, Z. Guo and J. Wang, *Adv. Energy Mater.*, 2021, **11**, 2100957.
- 39 J. Holoubek, H. Liu, Z. Wu, Y. Yin, X. Xing, G. Cai, S. Yu, H. Zhou, T. A. Pascal, Z. Chen and P. Liu, Nat. Energy, 2021, 6, 303–313.

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- 40 R. Xu, S. Zhang, X. Shen, N. Yao, J. Ding, Y. Xiao, L. Xu, C. Yan and J. Huang, *Small Structures*, 2023, **4**, 2200400.
- 41 G. Cai, J. Holoubek, M. Li, H. Gao, Y. Yin, S. Yu, H. Liu, T. A. Pascal, P. Liu and Z. Chen, Proc. Natl. Acad. Sci., 2022, 119, e2200392119.
- 42 S. Chen, J. Fan, Z. Cui, L. Tan, D. Ruan, X. Zhao, J. Jiang, S. Jiao and X. Ren, *Angew. Chem. Int. Ed.*, 2023, **62**, e202219310.
- 43 S.G. Yoon, K.A. Cavallaro, B.J. Park, H. Yook, J.W. Han and M.T. McDowell, *Adv. Funct. Mater.*, 2023, **33**, 2302778.
- 44 P. Liang, H. Hu, Y. Dong, Z. Wang, K. Liu, G. Ding and F. Cheng, Adv. Funct. Mater., 2024, 34, 2309858.
- 45 L. Cheng, Y. Wang, J. Yang, M. Tang, C. Zhang, Q. Zhu, S. Wang, Y. Li, P. Hu and H. Wang, Adv. Funct. Mater., 2022, 33, 2212349.
- 46 K. Naoi, E. Iwama, N. Ogihara, Y. Nakamura, H. Segawa and Y. Ino, J. Electrochem. Soc., 2009, **156**, A272.
- 47 Z. Liu, R. Wang, Q. Ma, H. Kang, L. Zhang, T. Zhou and C. Zhang, Carbon Neutralization, 2022, 1, 126–139.
- 48 J. Cuan, Y. Zhou, T. Zhou, S. Ling, K. Rui, Z. Guo, H. Liu and X. Yu, *Adv. Mater.*, 2019, **31**, 1803533.
- 49 M. Jiang, F. Zhang, G. Zhu, Y. Ma, W. Luo, T. Zhou and J. Yang, ACS Appl. Mater. Interfaces, 2020, 12, 24796–24805.
- L. Luo, K. Chen, H. Chen, H. Li, R. Cao, X. Feng, W. Chen Y. Fang,
 Y. Cao, Adv. Mater., 2023, 36, 2308881.
- 51 T. Zheng, J. Xiong, B. Zhu, X. Shi, Y.-J. Cheng, H. Zhao and Y. Xia, *J. Mater. Chem. A*, 2021, **9**, 9307–9318.
- 52 D. Lu, R. Li, M. M. Rahman, P. Yu, L. Lv, S. Yang, Y. Huang, C. Sun, S. Zhang, H. Zhang, J. Zhang, X. Xiao, T. Deng, L. Fan, L. Chen, J. Wang, E. Hu, C. Wang and X. Fan, *Nature*, 2024, 627, 101–107.
- 53 N. Von Aspern, G. -V. Röschenthaler, M. Winter and I. Cekic-Laskovic, *Angew. Chem. Int. Ed.*, 2019, **58**, 15978–16000.
- 54 H. Li, R. Hua, Y. Xu, D. Ke, C. Yang, Q. Ma, L. Zhang, T. Zhou and C. Zhang, *Chem. Sci.*, 2023, **14**, 10147–10154.
- 55 Y. Zou, Z. Ma, G. Liu, Q. Li, D. Yin, X. Shi, Z. Cao, Z. Tian, H. Kim, Y. Guo, C. Sun, L. Cavallo, L. Wang, H. N. Alshareef, Y. Sun and J. Ming, *Angew. Chem. Int. Ed.*, 2023, **135**, e202216189.
- 56 Y. Yang, Z. Fang, Y. Yin, Y. Cao, Y. Wang, X. Dong and Y. Xia, Angew. Chem. Int. Ed., 2022, 61, e202208345.
- 57 Y. Yang, P. Li, N. Wang, Z. Fang, C. Wang, X. Dong and Y. Xia, Chem. Commun., 2020, 56, 9640–9643.
- 58 Z. Cui, C. Liu and A. Manthiram, Adv. Mater., 2024, 36, 2409272.
- 59 Y. Mo, G. Liu, Y. Yin, M. Tao, J. Chen, Y. Peng, Y. Wang, Y. Yang, C. Wang, X. Dong and Y. Xia, *Adv. Energy Mater.*,2023, **13**, 2301285.
- 60 Z. Wang, Z. Sun, Y. Shi, F. Qi, X. Gao, H. Yang, H.-M. Cheng and F. Li, *Adv. Energy Mater.*, 2021, **11**, 2100935.
- 61 J. Xu, J. Zhang, T. P. Pollard, Q. Li, S. Tan, S. Hou, H. Wan, F. Chen, H. He, E. Hu, K. Xu, X. Yang, O. Borodin and C. Wang, *Nature*, 2023, **614**, 694–700.
- 62 P. Xiao, Y. Zhao, Z. Piao, B. Li, G. Zhou and H. Cheng, *Energy Environ. Sci.*, 2022, **15**, 2435–2444.
- 63 Y. Yang, W. Yang, H. Yang and H. Zhou, eScience, 2023, 3, 100170.
- 64 R. Wang, H. Wang, H. Zhao, M. Yuan, Z. Liu, G. Zhang, T. Zhang, Y. Qian, J. Wang, I. Lynch and Y. Deng, *Energy Mater.*, 2023, 3, 300040.
- 65 G. Zhang, J. Chang, L. Wang, J. Li, C. Wang, R. Wang, G. Shi, K. Yu, W. Huang, H. Zheng, T. Wu, Y. Deng and J. Lu, *Nat. Commun.*, 2023, **14**, 1081.
- 66 Y. Xue, Y. Wang, H. Zhang, W. Kong, Y. Zhou, B. Kang, Z. Huang and H. Xiang, *Angew. Chem. Int. Ed.*, 2024, **137**, e202414201.
- 67 L. Tan, P. Chen, Q. Chen, X. Huang, K. Zou, Y. Nie and L. Li, *Rare Met.*, 2023, 42, 4081–4090.
- 68 L. Jiang, C. Yan, Y. Yao, W. Cai, J. Huang and Q. Zhang, *Angew. Chem. Int. Ed.*, 2021, **60**, 3402–3406.

- 69 X. Zhang, X. Chen, X. Cheng, B. Li, X. Shen, C. Yan, J. Huang and Q. Zhang, *Angew. Chem. Int. Ed.*, 2018, 57, 530155392000035A
- 70 N. Xin, Y. Sun, M. He, C. J. Radke and J. M. Prausnitz, *Fluid Phase Equilib.*, 2018, **461**, 1–7.
- 71 X. Xu, X. Yue, Y. Chen and Z. Liang, *Angew. Chem. Int. Ed.*, 2023, **62**, e202306963.
- 72 P. Xiao, R. Luo, Z. Piao, C. Li, J. Wang, K. Yu, G. Zhou and H.M. Cheng, *ACS Energy Lett.*, 2021, **6**, 3170–3179.
- 73 H. Chen, K. Chen, L. Luo, X. Liu, Z. Wang, A. Zhao, H. Li, X. Ai, Y. Fang and Y. Cao, Angew. Chem. Int. Ed., 2024, 63, e202316966.
- 74 Z. Jiang, T. Yang, C. Li, J. Zou, H. Yang, Q. Zhang and Y. Li, *Adv. Funct. Mater.*, 2023, **33**, 2306868.
- 75 R. Yu, Z. Li, X. Zhang and X. Guo, *Chem. Commun.*, 2022, **58**, 8994–8997.
- 76 F. Qiu, X. Li, H. Deng, D. Wang, X. Mu, P. He and H. Zhou, Adv. Energy Mater., 2019, 9, 1803372.
- 77 N. Piao, S. Liu, B. Zhang, X. Ji, X. Fan, L. Wang, P. Wang, T. Jin, S. Liou, H. Yang, J. Jiang, K. Xu, M. A. Schroeder, X. He and C. Wang, *ACS Energy Lett.*, 2021, **6**, 1839–1848.
- 78 Q. Zhang, S. Sun, M. Zhou, L. Hou, J. Liang, S. Yang, B. Li, X. Zhang and J. Huang, *Angew. Chem. Int. Ed.*, 2023, **135**, e202306889.
- 79 R. Zhao, X. Li, Y. Si, W. Guo and Y. Fu, *ACS Appl. Mater. Interfaces*, 2021, **13**, 40582–40589.
- 80 A. C. Thenuwara, P. P. Shetty, N. Kondekar, S. E. Sandoval, K. Cavallaro, R. May, C. Yang, L. E. Marbella, Y. Qi and M. T. McDowell, *ACS Energy Lett.*, 2020, **5**, 2411–2420.
- 81 D. Zhang, D. Zhu, W. Guo, C. Deng, Q. Xu, H. Li and Y. Min, Adv Funct. Mater., 2022, 32, 2112764.
- 82 J. Chen, Y. Zhang, H. Lu, J. Ding, X. Wang, Y. Huang, H. Ma and J. Wang, *eScience*, 2023, **3**, 100135.
- 83 X. Lan, S. Yang, T. Meng, C. Zhang and X. Hu, *Adv. Energy Mater.*, 2023, **13**, 2203449.
- 84 G. Kang, G. Zhong, K. Cai, J. Ma, J. Biao, Y. Cao, S. Lu, K. Yu, F. Kang and Y. Cao, *ACS Energy Lett.*, 2024, **9**, 2572–2581.
- 85 B. Liao, H. Li, M. Xu, L. Xing, Y. Liao, X. Ren, W. Fan, L. Yu, K. Xu and W. Li, *Adv. Energy Mater.*,2018, **8**, 1800802.
- 86 B. Yang, H. Zhang, L. Yu, W. Fan and D. Huang, *Electrochim. Acta*, 2016, **221**, 107–114.
- 87 G. Xu, S. Huang, Z. Cui, X. Du, X. Wang, D. Lu, X. Shangguan, J. Ma, P. Han, X. Zhou and G. Cui, *J. Power Sources*, 2019, **416**, 29–36.
- 88 H. Jiang, C. Yang, M. Chen, X. Liu, L. Yin, Y. You and J. Lu, *Angew. Chem. Int. Ed.*, 2023, **62**, e202300238.
- 89 Y. Lin, X. Yue, H. Zhang, L. Yu, W. Fan and T. Xie, *Electrochimica Acta*, 2019, **300**, 202–207.
- 90 S. Jurng, S. Park, T. Yoon, H. Kim, H. Jeong, J. H. Ryu, J. J. Kim and S. M. Oh, *J. Electrochem. Soc.*, 2016, **163**, A1798–A1804.
- 91 Y. Yao, X. Chen, C. Yan, X. Zhang, W. Cai, J. Huang and Q. Zhang, *Angew. Chem. Int. Ed.*, 2021, **60**, 4090–4097.
- 92 H. Zhang, Z. Zeng, F. Ma, Q. Wu, X. Wang, S. Cheng and J. Xie, Angew. Chem. Int. Ed., 2023, 135, e202300771.
- 93 J. Wu, Z. Gao, Y. Tian, Y. Zhao, Y. Lin, K. Wang, H. Guo, Y. Pan, X. Wang, F. Kang, N. Tavajohi, X. Fan and B. Li, *Adv. Mater.*, 2023, **35**, 2303347.
- 94 Y. Zhao, Z. Hu, Z. Zhao, X. Chen, S. Zhang, J. Gao and J. Luo, *J. Am. Chem. Soc.*, 2023, **145**, 22184–22193.
- 95 J. Wu, S. Zhang, C. Yang, X. Zhang, M. Zhou, W. Liu and H. Zhou, *Energy Stor. Mater.*, 2023, **63**, 103043.
- 96 F. Cheng, W. Zhang, Q. Li, C. Fang, J. Han and Y. Huang, ACS Nano, 2023, 17, 24259-24267.
- 97 L. Chen, J. Wang, M. Chen, Z. Pan, Y. Ding, Z. Song, X. Ai, Y. Cao and Z. Chen, *Energy Stor. Mater.*, 2024, **65**, 103098.
- 98 Q. Ma, J, Zheng, H. Kang, L. Zhang, Q. Zhang, H. Li, R. Wang, T. Zhou, Q. Chen, A. Liu, H. Li and C. Zhang, ACS Appl. Mater. Interfaces, 2021, 13, 43002–43010.

ES Batteries Accepted Manuscript

COMMUNICATION Journal Name

- 99 L. Liu, Z. Shadike, N. Wang, Y. Chen, X. Cai, E. Hu and J. Zhang, eScience, 2024, 3, 100268.
- L. Liu, Z. Shadike, N. Wang, Y. Chen, X. Cai, E. Hu and J, Zhang, eScience, 2024, 4, 100268
- Z. Wang, H. Zhang, J. Xu, A. Pan, F. Zhang, L. Wang, R. Han, J. Hu, M. Liu and X. Wu, Adv. Funct. Mater., 2022, 32, 2112598.
- J. Wang, Q. Zheng, M. Fang, S. Ko, Y. Yamada and A. Yamada, Adv. Sci., 2021, 8, 2101646.
- X. Cao, H. Jia, W. Xu and J.-G. Zhang, J. Electrochem. Soc., 2021, 168, 010522.
- J. Cuan, Y. Zhou, T. Zhou, S. Ling, K. Rui, Z. Guo, H. Liu and X. Yu, Adv. Mater., 2019, 31, 1803533.
- H. Wu, X. Zhou, C. Yang, D. Xu, Y.-H. Zhu, T. Zhou, S. Xin and Y. You, ACS Appl. Mater. Interfaces, 2023, 15, 18828-18835.
- H. Jia, J. Kim, P. Gao, Y. Xu, M. H. Engelhard, B. E. Matthews, C. Wang and W. Xu, Angew. Chem. Int. Ed., 2023, 135, e202218005.
- H. Wang, J. Liu, J. He, S. Qi, M. Wu, F. Li, J. Huang, Y. Huang and J. Ma, eScience, 2022, 2, 557-565.
- J. Duan, H. Pei, Q. Yang, X. Li, X. Ba, X. Yong, J. Guo and S. Lu, Rare Met., 2024, 43, 2560-2573.
- J. Chen, H. Zhang, M. Fang, C. Ke, S. Liu and J. Wang, ACS Energy Lett., 2023, 8, 1723-1734.
- J. Holoubek, K. Kim, Y. Yin, Z. Wu, H. Liu, M. Li, A. Chen, H. Gao, G. Cai, T. A. Pascal, P. Liu and Z. Chen, Energy Environ. Sci., 2022, 15, 1647-1658.
- S. Lin, H. Hua, P. Lai and J. Zhao, Adv. Energy Mater., 2021, **11,** 2101775.
- X. Zhang, L. Zou, Y. Xu, X. Cao, M. H. Engelhard, B. E. Matthews, L. Zhong, H. Wu, H. Jia, X. Ren, P. Gao, Z. Chen, Y. Qin, C. Kompella, B. W. Arey, J. Li, D. Wang, C. Wang, J. Zhang and W. Xu, Adv. Energy Mater., 2020, 10, 2000368.
- N. Piao, J. Wang, X. Gao, R. Li, H. Zhang, G. Hu, Z. Sun, X. Fan, H. Cheng and F. Li, J. Am. Chem. Soc., 2024, 146, 18281-
- J. Shi, C. Xu, J. Lai, Z. Li, Y. Zhang, Y. Liu, K. Ding, Y. Cai, R. Shang and Q. Zheng, Angew. Chem. Int. Ed., 2023, 135, e202218151.
- X. Dong, Y. Lin, P. Li, Y. Ma, J. Huang, D. Bin, Y. Wang, Y. Qi and Y. Xia, Angew. Chem. Int. Ed., 2019, 58, 5623-5627.
- X. Fan, X. Ji, L. Chen, J. Chen, T. Deng, F. Han, J. Yue, N. Piao, R. Wang, X. Zhou, X. Xiao, L. Chen and C. Wang, Nat. Energy, 2019, 4, 882-890.
- Z. Cui, D. Wang, J. Guo, Q. Nian, D. Ruan, J. Fan, J. Ma, L. Li, Q. Dong, X. Luo, Z. Wang, X. Ou, R. Cao, S. Jiao and X. Ren, J. Am. Chem. Soc., 2024, 146, 27644-27654.
- X. Liu, A. Mariani, T. Diemant, X. Dong, P. Su and S. Passerini, Angew. Chem. Int. Ed., 2023, 62, e202305840.
- C. S. Rustomji, Y. Yang, T. K. Kim, J. Mac, Y. J. Kim, E. Caldwell, H. Chung and S. Meng, Science, 2017, 356, eaal4263.
- Y. Y. C. Yang, D. M. Davies, Y. J. Yin, O. Borodin, J. Z. Lee, C. C. Fang, M. Olguin, Y. H. Zhang, E. S. Sablina, X. F. Wang, C. S. Rustomji and Y. S. Meng, Joule, 2019, 3, 1986-2000.
- Y. Yang, Y. Yin, D. M. Davies, M. Zhang, M. Mayer, Y. Zhang, E. S. Sablina, S. Wang, J. Z. Lee, O. Borodin, C. S. Rustomji and Y. S. Meng, Energy Environ. Sci., 2020, 13, 2209-2219.
- Y. Yin, Y. Yang, D. Cheng, M. Mayer, J. Holoubek, W. Li, G. Raghavendran, A. Liu, B. Lu, D. D, Z. Chen, O. Borodin and Y. Meng, Nat. Energy, 2022, 7, 548-559.
- Z. Li, Y. Yao, M. Zheng, S. Sun, Y. Yang, Y. Xiao, L. Xu, C. Jin, X. Yue, T. Song, P. Wu, C. Yan and Q. Zhang, Angew. Chem. Int. Ed., 2024, 137, e202409409.
- A. Dave, J. Mitchell, S. Burke, H. Lin, J. Whitacre and V. Viswanathan, Nat. Commun., 2022, 13, 5454.

- E. Miele, W. M. Dose, I. Manyakin, M. H. Frosz, Z. Ruff, M. F. L. De Volder, C. P. Grey, J. J. Baumberg and To Go Euser Mat Commun., 2022, 13, 1651.
- K. Qian, Z. Yu, Y. Liu, D. Gosztola, R. Winans, L. Cheng and T. Li, J. Energy Chem., 2022, 70, 340-346.
- X. Zhou, Y. Zhou, L. Yu, L. Qi, K.-S. Oh, P. Hu, S.-Y. Lee and C. Chen, Chem. Soc. Rev., 2024, 53, 5291-5337.
- X. Hu, Y. Zheng, D.A. Howey, H. Perez, A. Foley and M. Pecht, Prog. Energy and Combustion Sci., 2020, 77, 100806.

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