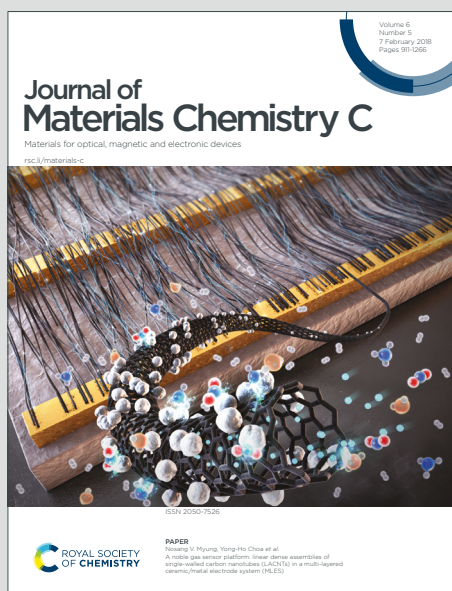


Journal of Materials Chemistry C

Materials for optical, magnetic and electronic devices

Accepted Manuscript

This article can be cited before page numbers have been issued, to do this please use: D. Gochitashvili, C. R. Tomassetti, E. R. Margine and A. Kolmogorov, *J. Mater. Chem. C*, 2025, DOI: 10.1039/D5TC02237A.



This is an Accepted Manuscript, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about Accepted Manuscripts in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this Accepted Manuscript or any consequences arising from the use of any information it contains.

High- T_c Ag $_x$ BC and Cu $_x$ BC superconductors accessible via topochemical reactionsDaviti Gochitashvili,¹ Charley R. Tomassetti,¹ Elena R. Margine,¹ and Aleksey N. Kolmogorov^{1,*}¹Department of Physics, Applied Physics, and Astronomy,
Binghamton University-SUNY, Binghamton, New York 13902, USA

(Dated: July 17, 2025)

Hole-doping of covalent materials has long served as a blueprint for designing conventional high- T_c superconductors, but thermodynamic constraints severely limit the space of realizable compounds. Our *ab initio* results indicate that metastable Ag $_x$ BC and Cu $_x$ BC phases can be accessed via standard topochemical ion exchange reactions starting from Li $_x$ BC precursors. Unlike all known stoichiometric layered metal borocarbides, the predicted AgBC and CuBC derivatives, comprising honeycomb layers bridged by dumbbells, are metallic rather than semiconducting. Anisotropic Migdal-Eliashberg analysis reveals that the intrinsically hole-doped AgBC possesses a unique combination of electronic and vibrational features to exhibit two-gap superconductivity above 50 K.

I. Introduction

The search for new high- T_c superconductors has yielded a large number of theoretical predictions but few experimental validations [1, 2]. This disconnect reflects an inverse correlation between thermodynamic stability and conventional superconductivity, now established across different material classes [3–7]. With the vast majority of proposed high- T_c superconductors being, at best, metastable at ambient conditions, high-pressure synthesis with possible quenching has enabled key discoveries [8, 9] but remains impractical for scalable applications. Therefore, predictive efforts must address synthesis pathways alongside superconductor design, and soft chemistry methods offer a promising route to bypass thermodynamic constraints.

Topochemical reactions, first reported in 1931 [10, 11], provide access to (meta)stable crystalline materials without the need for extreme temperature or pressure synthesis conditions. The selective replacement or removal of species at specific atomic sites while leaving the structural backbone essentially unchanged represents a kinetically controlled pathway for compositional modifications. Over the past decades, the approach has been used to produce a variety of new layered compounds, including metal oxides [12–16], (oxy)chalcogenides [17–20], and borides [21–23]. The resulting materials have exhibited enhanced photocatalytic activity [12], potential for spin-liquid behavior [24], improved electrochemical performance [23], and unconventional superconductivity [25].

The extensive experimental work has identified several factors governing topochemical transformations, including ion size mismatch, reactant selection, and reaction environment. Topotactic ion exchange preserves the original crystal structure and typically favors the replacement of larger cations with smaller ones due to diffusion kinetics and lattice constraints, as seen in Cu substituting Na in Na₂IrO₃ [14, 15], but the reverse

has also been observed, such as Ag substituting Li in α -Li₂IrO₃ [24, 26]. Some reactions involve rich chemistry with multiple stages [20, 21, 27], while targeted ion exchange is often facilitated by the introduction of cations in the form of metal salts [12, 18]. *Ab initio* work has focused primarily on interpreting topochemical pathways or modeling the observed materials' properties [13, 28, 29]. Moving toward *ab initio* materials design, Suzuki et al. [30] recently introduced a computational framework for identifying feasible topotactic reactions and predicted ternary wurtzite-type oxides by calculating free energies with density functional theory (DFT). Their experimental work validated the DFT-derived trends, leading to previously unreported Li-to-Ag exchange and the synthesis of multiple metastable compounds and solid solutions with controlled compositions.

The present study focuses on predicting new layered metal borocarbides, a diverse materials class unexplored with topochemical ion exchange. Over the years, LiBC [31] has attracted wide attention due to early theoretical predictions suggesting that its hole-doped derivatives could exhibit high-temperature MgB₂-type superconductivity resulting from the coupling of the in-plane BC- $p_{x,y}$ orbitals and the bond-stretching modes of the electron-deficient BC framework [32, 33]. However, despite extensive experimental efforts, superconductivity has not been observed in delithiated LiBC [34–39], likely due to deviations from the ideal BC morphology arising from the removal of Li [40]. Namely, our *ab initio* calculations showed that layer buckling and interlayer bridging reduce the T_c by factors of ~ 2 and ~ 20 , respectively, while the intralayer B-C bond rotation suppresses superconductivity completely. Our following DFT analysis indicated the possibility of modulating the structure through soft chemical techniques, such as thermodynamically favorable reintercalation of Li $_x$ BC with alkaline or alkaline earth metals [40], obtaining mixed-metal Li $_x$ M $_y$ BC materials with signature electronic and vibrational features requisite for high- T_c superconductivity. Beyond LiBC, only two other metal borocarbides with honeycomb layers have been reported, MgB₂C₂ [41], and BeB₂C₂ [42], with another, NaBC [40], previously

* kolmogorov@binghamton.edu



predicted as a low-temperature ground state. Our *ab initio* phase stability and electron-phonon (e-ph) coupling calculations revealed that thermal deintercalation of MgB_2C_2 and NaBC under accessible experimental conditions could turn the semiconducting materials into hole-doped superconductors with T_c values in the 43–84 K range [43]. These proposed deintercalation or reintercalation pathways rely on the robustness of the BC layered framework, as LiBC has been shown to retain its honeycomb morphology at the ideal 1:1 stoichiometry at high temperatures (up to 1770 K) and low metal concentrations (down to at least $x = 0.5$) [38, 44].

Here, we explore the kinetically protected subspace of borocarbides with honeycomb layers through topochemical reactions facilitated by the introduction of metal salts. Our DFT results show that the substitution of alkali or alkaline earth metals in known borocarbides with Cu, Ag, or Zn releases ~ 100 kJ/mol, comparable to the calculated reaction energy in successful topochemical syntheses of delafossites [15, 30]. We carried out extensive structure searches to identify low-energy layered Li-Cu-BC and Li-Ag-BC derivatives and analyzed the superconducting properties of AgBC with the anisotropic Migdal-Eliashberg (aME) formalism [45, 46]. Fermi surfaces from the signature hole-doped BC- $p_{x,y}$ states, electron-doped Ag- s , and BC- p_z states in the stoichiometric AgBC are found to boost the T_c above 50 K. The findings suggest a promising route to a new class of ternary and, possibly, quaternary metal borocarbide superconductors with high T_c .

II. Methods

To analyze stability of M-B-C and Li-M-BC phases we employed VASP [47] using projector augmented wave potentials [48] and a 500 eV plane-wave cutoff. As the subject of the study was layered materials, where dispersive interactions are important [49–52], we relied on the optB86b-vdW functional [53]. Additionally, we reexamined the stability of the final set of predicted phases using $r^2\text{SCAN}+r\text{VV10}$ [52, 54], and employed HSE06 [55] to obtain more reliable band gap predictions for selected phases. All structures were evaluated with dense ($\Delta k \sim 2\pi \times 0.025 \text{ \AA}^{-1}$) Monkhorst-Pack $\mathbf{k}$ -meshes [56]. Thermodynamic stability values are reported primarily in eV per atom of the product, such as AgBC with three atoms per formula unit, while reaction energies are also given in kJ per mole to facilitate the comparison with previous studies (see Supplementary Note I).

Global structure optimizations were performed using an evolutionary algorithm implemented in the MAISE package [57]. In fixed-composition runs, randomly initialized 16-member populations with up to 22 atoms per unit cell evolved for up to 250 generations through standard mutation and crossover operations [57]. For the systematic screening of possible interlayer site decorations in metal borocarbides, we constructed various CuBC and

AgBC cells and sequentially substituted and/or removed metals while retaining only the nonequivalent configurations. Equivalence was determined using our structural fingerprint based on the radial distribution function [57, 58]. The generated supercells included standard expansions of the fully occupied hP6 structure, recognized as the ground state for LiBC [31]. These expansions comprised $\sqrt{3} \times \sqrt{3} \times 1$, $2 \times 2 \times 1$, and larger orthorhombic supercells with up to 54 atoms. We considered both AA and AA' stackings of the BC layers. In total, we examined a diverse set of layered phases, with more than 4000 unique metal arrangements for each system.

For selected structures, the thermodynamic corrections due to vibrational entropy were evaluated using the finite displacement method implemented in PHONOPY [59]. We used supercells ranging from 69 to 264 atoms, with an average of 138 atoms across 80 phases, and applied 0.1 Å displacements within the harmonic approximation. *Ab initio* molecular dynamics (AIMD) simulations were performed using the VASP code, with a 400 eV energy plane-wave cutoff and $4 \times 4 \times 2$ $\mathbf{k}$ -mesh for the $3 \times 3 \times 2$ supercell, to evaluate the thermal stability of the predicted AgBC phase. The simulations were carried out in the *NVT* ensemble using a Nosé-Hoover thermostat at a constant temperature of 600 K for 10,000 1-fs timesteps.

The QUANTUM ESPRESSO package [60] was employed for calculating ground-state properties associated with superconductivity, using the optB86b-vdW [53] functional and norm-conserving pseudopotentials from the Pseudo Dojo library [61]. The lattice parameters and atomic positions were relaxed until the total energy was converged within 10^{-6} Ry and the maximum force on each atom was less than 10^{-4} Ry/Å. A plane-wave cutoff value of 100 Ry, a Methfessel-Paxton smearing [62] value of 0.02 Ry, and Γ -centered Monkhorst-Pack [56] $\mathbf{k}$ -meshes were used to describe the electronic structure. The dynamical matrices, the linear variation of the self-consistent potential, and electron-phonon coupling strength were calculated within density-functional perturbation theory [63] on irreducible sets of regular $\mathbf{q}$ -meshes given in Supplementary Table 3, shown in previous studies to ensure convergence for related layered materials [40, 43, 64]. For example, a $30 \times 30 \times 18$ $\mathbf{k}$ - and $6 \times 6 \times 4$ $\mathbf{q}$ -mesh were used for AgBC.

We utilized the EPW code [45, 65–67] to evaluate the e-ph interaction and superconducting properties of AgBC. The Wannier interpolation [68–70] was performed on a uniform $6 \times 6 \times 4$ Γ -centered $\mathbf{k}$ -grid with the Wannier90 code [68–70] in library mode. As projections for the maximally localized Wannier functions, we used s and p orbitals for every C atom, and d orbitals for every Ag atom to reproduce the electronic structure, totaling 9 Wannier functions. The anisotropic full-bandwidth ME equations [46, 67] were solved with a sparse intermediate representation of the Matsubara frequencies [71] on fine uniform $100 \times 100 \times 70$ $\mathbf{k}$ - and $50 \times 50 \times 35$ $\mathbf{q}$ -point grids, with an energy window of ± 0.4 eV around the Fermi level. The Coulomb μ^* parameter was chosen to be 0.20



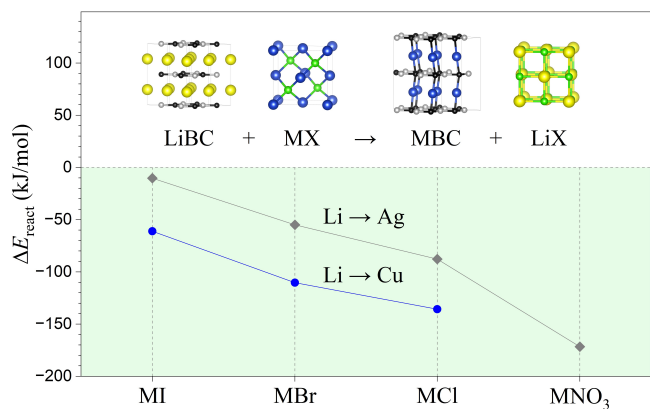


FIG. 1. Energy per mole for the topochemical reactions between stoichiometric LiBC and metal salts XM ($X = \text{I, Br, Cl, or NO}_3$) resulting in the formation of MBC ($M = \text{Cu or Ag}$). Negative energies indicate exothermic reactions.

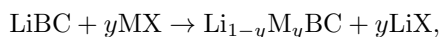
in order to ensure agreement between measured and computed aME T_c values for MgB_2 [64]. The EPW code was also used to find the isotropic Migdal-Eliashberg critical temperatures using the Eliashberg spectral functions outputted by the PHonon code of QUANTUM ESPRESSO.

Visual representations of crystal structures and Fermi surfaces were respectively created with VESTA [72] and FermiSurfer [73]. To resolve different phases at the same composition, we use Pearson symbols and space groups when needed. Full structural information for relevant DFT-optimized M_yBC or $\text{Li}_{x-y}\text{M}_y\text{BC}$ ($M = \text{Cu or Ag}$) phases is provided as CIF files in the Supplementary Material.

III. Results and discussion

A. Topochemical reaction energetics

Unlike conventional reactions that drive materials toward their most thermodynamically stable phases, topochemical transformations yield metastable products that are most favorable within the kinetically constrained space defined by precursors. The success of topochemical ion exchange in several classes of layered materials suggests that a similar process should be viable for metal borocarbides with strong honeycomb frameworks provided that the net Gibbs free energy change is negative. For example, a partial replacement of Li with another metal induced by reacting LiBC with the corresponding metal halide is described by



where X is Cl, Br, or I, and M is Na, Cu, or Ag. Suzuki et al. [30] observed full ion exchange for all reactions with DFT $\Delta E_{\text{react}} < -7.3$ kJ/mol (-0.025 eV/atom, see Supplementary Note I for unit conversion) and attributed the formation of disordered (Ag,Na)GaO₂ alloys with the

TABLE I. Calculated topochemical reaction energies in kJ per mole for different precursors and ion sources. The energetics of ion swaps in metal iridate [15] and gallate [30] compounds serve as a reference.

Precursor compound	Derived compound	ΔE_{react} (kJ/mol)				
		M	MI	MBr	MCl	MNO ₃
LiBC	NaBC	+100	+111	+104	+96	+80
LiBC	CuBC	+158	-61	-110	-136	
LiBC	AgBC	+197	-10	-55	-88	-172
BeB_2C_2	ZnB_2C_2	+116	+114	+209	+21	
MgB_2C_2	ZnB_2C_2	+142	-14	-58	-91	-185
Na_2IrO_3	Cu_2IrO_3	+214	-17	-59	-77	
LiGaO_2	AgGaO_2	+165	+142	+98	+65	-19

parent structure to the $-TS_{\text{conf}}$ configurational entropy contribution.

In the present DFT evaluation of reaction energies, we used structures with honeycomb BC layers and the most stable metal arrangements found in extensive searches detailed in Section III B. The most favorable mP6-CuBC derivative, for example, turned out to be 45 kJ/mol (0.157 eV/atom) lower in energy than the starting hP6-LiBC prototype. The energy difference is significantly higher than the free energy corrections from configurational and vibrational entropy discussed in Section III B, which highlights the importance of exploring the constrained space. The ground state structures for the known materials, taken from the Materials Project [74], are listed in the Supplementary Material.

The DFT ΔE_{react} values summarized in Fig. 1 and Table I indicate that full ion exchange is a downhill reaction for several precursor-salt combinations. The monovalent Li could be replaced with either Cu or Ag introduced as any of the considered halides. The Li-to-Cu ion exchange should be easier to achieve kinetically due to their similar sizes and a moderate 13% expansion of the interlayer spacing in the predicted mP6-CuBC. The substantial 32% interlayer expansion needed to accommodate Ag in the predicted hP3-AgBC and the less favorable energetics, by ~ 50 kJ/mol relative to CuBC for each salt, may hinder the reaction rate. $\Delta E_{\text{react}} = -10$ kJ/mol for the least exothermic $\text{LiBC} + \text{AgI}$ is comparable to values associated with the formation of mixed-metal products [30]. Our calculations show that the thermodynamics could be improved significantly, to -172 kJ/mol, by switching to an alternative AgNO_3 ion source, but limiting the energy release by using AgI/AgBr mixtures with optimized ratios could increase the chances of preserving the BC morphology.

Among the known alkaline-earth metal borocarbides, BeB_2C_2 has an oP20 ground state with differently stacked BC honeycomb layers [42], MgB_2C_2 crystallizes in a complex oS80 structure with BC honeycomb layers sandwiching Mg rhombus-shaped patches [41], and CaB_2C_2 adopts a tI20 structure with 4-8-polygon BC



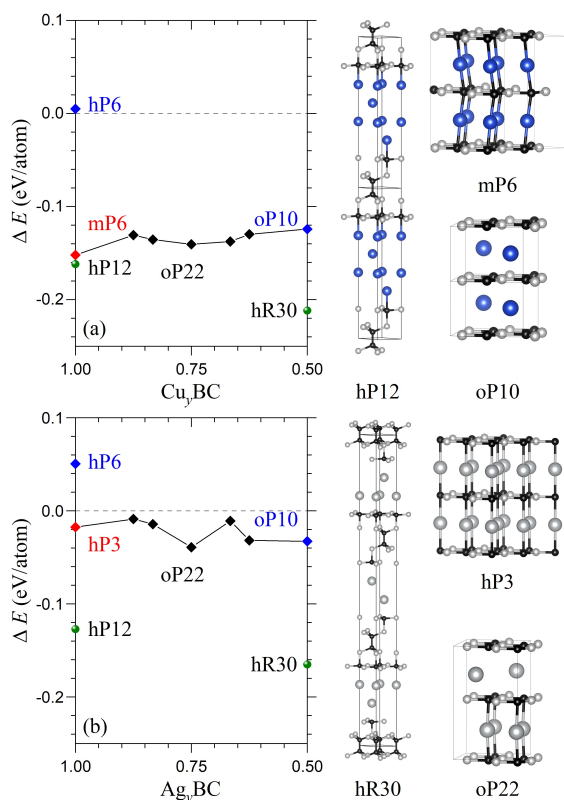


FIG. 2. Relative energies of Cu_xBC and Ag_xBC phases referenced to a hypothetical fully deintercalated BC phase and the elemental ground states. The line connects the lowest-energy phases with the honeycomb layer morphology. Both metals prefer to occupy dumbbell sites at $x = 1$ (red points) but shift to interstitial sites at $x = 1/2$ (blue points). The green points correspond to alternative motifs with bridged layers. The B, C, Cu, and Ag atoms in the displayed crystal structures are shown with light gray, black, blue, and gray spheres, respectively.

layers [75]. Our global structure searches indicate that ZnB_2C_2 would be isostructural to BeB_2C_2 and could be synthesized via topochemical Mg-to-Zn exchange leading to a minor 2% interlayer spacing reduction. Reactions with the proposed NaBC precursor, shown in Supplementary Table I, are even more exothermic. Given the promise of this material, we revisited the ternary system by considering two other stable Na-B-C phases recently entered into the Materials Project database, $\text{NaB}_{13}\text{C}_2$ (mp-3293300), synthesized under pressure [76], and NaBC_7 (mp-3230146), a possible Na-intercalated B-doped graphite. According to our revised relative stability plots (Supplementary Fig. 1), NaBC remains a low-temperature ground state and, as such, may not be accessible through high-temperature synthesis from the elements.

To illustrate the level of reliability of the DFT-based predictions for metal borocarbides, Table I lists calculated ΔE_{react} for select metal oxides investigated in previous studies. The large negative -77 kJ/mol is con-

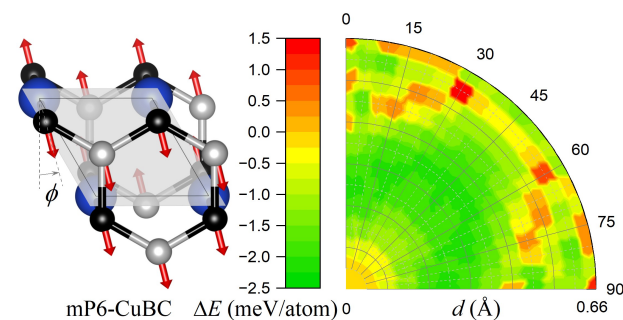


FIG. 3. Change in the CuBC energy as a function of inter-layer shift angle (ϕ) and magnitude (d) away from the hP3 configuration. The CuBC derivatives with the lowest mP6 symmetry were obtained by displacing adjacent BC layers in the $1 \times 1 \times 2$ hP3 supercell in the opposite directions.

sistent with the successful Na-to-Cu swap via $\text{Na}_2\text{IrO}_3 + \text{CuCl}$ [15]. Our calculated energy changes for the Li-to-Ag swap in delafossites agree within ~ 5 kJ/mol relative to reported values obtained with the PBEsol functional [30] and correctly identify AgNO_3 as the only salt enabling synthesis of AgGaO_2 . Notably, the calculated volume change in this topotactic reaction is fairly isotropic and reaches 26%.

Topochemical reactions discussed so far involve stoichiometric precursors but the demonstrated [34–39] or predicted [40, 43] thermal deintercalation of metal borocarbides offers a new dimension in the design of derived materials. Namely, while the direct replacement of the intercalant (Li, Be, or Mg) with a new metal (Na, Cu, Ag, or Zn) is disfavored for all known metal borocarbides listed in Table I, reintercalation of initially delithiated Li_xBC with different metals (Na, K, Mg, or Ca) has been shown to be thermodynamically favorable [40]. Hence, our present study also includes the screening of non-stoichiometric ternary M_{x-y}BC and mixed-metal quaternary $\text{Li}_{x-y}\text{M}_y\text{BC}$ derivatives with $\text{M} = \text{Cu}$ or Ag .

B. Thermodynamic stability

The exploration of the constrained $\text{Li}_{x-y}\text{M}_y\text{BC}$ ($1 \geq x \geq 1/2$ and $1 > x - y \geq 0$) phase space originating from the precursor composition relied on two complementary strategies. Combinatorial screening of supercells with honeycomb BC layers enabled identification of stable intercalant decorations, while global evolutionary searches helped establish favorable composition-dependent motifs beyond the parent morphology.

Our first evolutionary runs for ternary M_yBC compounds with $\text{M} = \text{Cu}$ or Ag uncovered a notable difference between preferred locations of the transition and alkali or alkaline-earth metals in the BC honeycomb matrices. In contrast to Li, Na, K, Mg, Ca, or their Li-M combinations that tend to be in or near the middle of the hexagons between AA'-stacked BC layers [40, 43],



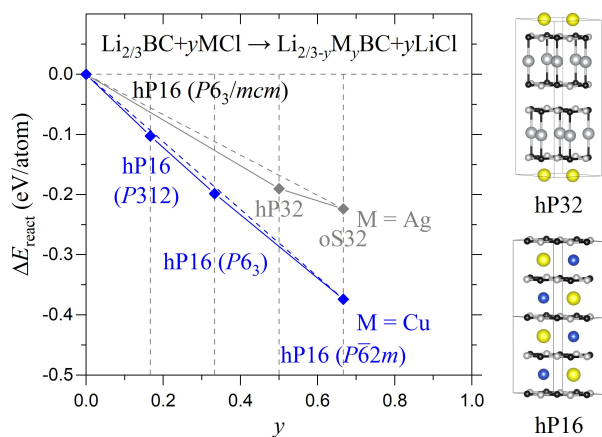


FIG. 4. Reaction energy per atom of topochemical ion exchange between $\text{Li}_{2/3}\text{BC}$ and $y\text{MCl}$ ($\text{M} = \text{Cu}$ or Ag). The solid points mark quaternary phases stable with respect to $\text{Li}_{2/3}\text{BC}$ and $\text{M}_{2/3}\text{BC}$ and forming the local convex hull, such as the displayed $\text{hP16-Li}_{1/3}\text{Cu}_{1/3}\text{BC}$ ($P6_3$) and $\text{hP32-Li}_{1/6}\text{Cu}_{1/2}\text{BC}$.

Cu and Ag showed a clear preference to form C-M-C dumbbells forcing the AA stacking sequence. Therefore, we generalized the decoration sampling protocol to allow initial positions of the transition metals to be either at the interstitial or dumbbell sites and imposed a constraint on the minimum distance between neighboring metals to avoid unphysical starting decorations. The adjustment expanded the number of unique states by over 1000 structures across the considered composition range. Figure 2 displays the lowest-energy phases at each composition for dumbbell, interstitial, or mixed decorations. The results show that the two ternary systems prefer to adopt dumbbell and interstitial configurations at the full and half-filled levels, respectively. Phases at intermediate stoichiometries, e.g., oP22 at $y = 3/4$, represent mixtures of fully-filled galleries with dumbbells and half-filled galleries with interstitials. The stoichiometric CuBC and AgBC , natural candidates to form with the LiBC precursor, warrant closer examination. AgBC is dynamically and thermally stable in the hP3 structure with linear C-Ag-C links, whereas hP3-CuBC exhibits imaginary phonon modes shearing the BC layers (see Supplementary Figs. 2 and 3 and Section III C). These observations align with the known tendency of both metals to form dumbbells, and with the electronic frustration observed in Cu -based materials, as Cu^{1+} often disproportionates into Cu^0 and Cu^{2+} [77]. Displacement of the atoms along the A-point imaginary phonon eigenmode in the doubled unit cell generated a dynamically stable mP6-CuBC phase 2.5 meV/atom lower in energy (Fig. 3). However, the presence of the radially degenerate minima around the hP3 stationary point shows that CuBC may have a complex structure with stacking disorder that will require further investigation. For example, the $\text{r}^2\text{SCAN+rVV10}$ functional, better suited to handle vari-

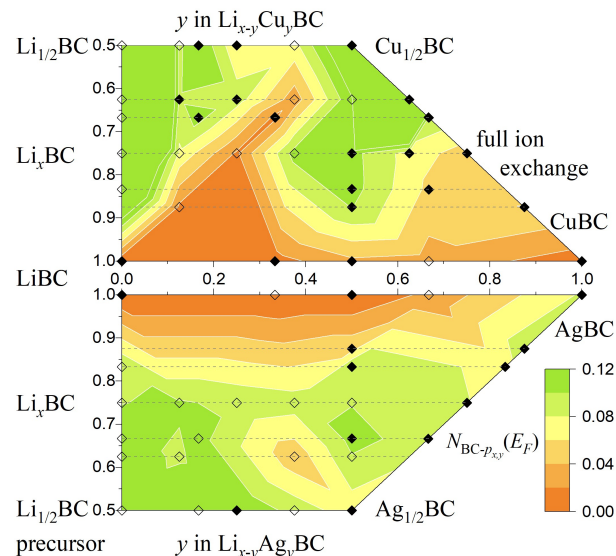


FIG. 5. Projected DOS from $\text{BC-}p_{x,y}$ (states/(eV atom)) in $\text{Li}_{x-y}\text{M}_y\text{BC}$ compounds with $\text{M} = \text{Cu}$ or Ag . The solid diamonds mark ternary and locally stable quaternary compounds at 600 K. The contour map is based on compounds within 25 meV/atom to the local convex hulls marked with solid or hollow diamonds.

able oxidation states and capture medium-range correlation [77], also produces an imaginary phonon mode at the A point but favors mP6 over hP3 by only 0.5 meV/atom. In general, predicting and resolving distorted configurations represent both computational and experimental challenges, with some of the proposed revisions, such as the lower-symmetry ground state of Cu_2IrO_3 , have yet to be confirmed [78]. The electronic structure analysis in Section III C sheds light on the origin and consequences of the dumbbell linkage.

Figure 2 demonstrates that insertion of the transition metals into a hypothetical empty honeycomb BC framework leads to significantly lower stabilization, below 0.15 eV/atom for CuBC and 0.01 eV/atom for AgBC , compared to $\sim 0.4 - 0.7$ eV/atom for LiBC , NaBC , or MgB_2C_2 [43, 44]. The metastable layered transition metal borocarbides are also further above the global convex hull facet of M , C , and B_4C , by at least 0.16 eV/atom for Cu_yBC and 0.3 eV/atom for Ag_yBC in the full investigated range of y . In fact, CuBC and AgBC have positive formation energies of 0.12 and 0.25 eV/atom, respectively, but the unfavorable energetics prohibiting the direct synthesis from the elements does not automatically disallow exothermic reactions steered by kinetics. For example, a related layered BC_3 , unstable with respect to B and C by a comparable 0.20 eV/atom [44], has been synthesized from different precursors [21, 79]. A more pertinent question is whether M_yBC compounds have polymorphs with derived morphologies, such as linked-layer structures favored in significantly delithiated LiBC [40], that could form instead or serve as transient points *en route* to full decomposition. Our evolutionary searches



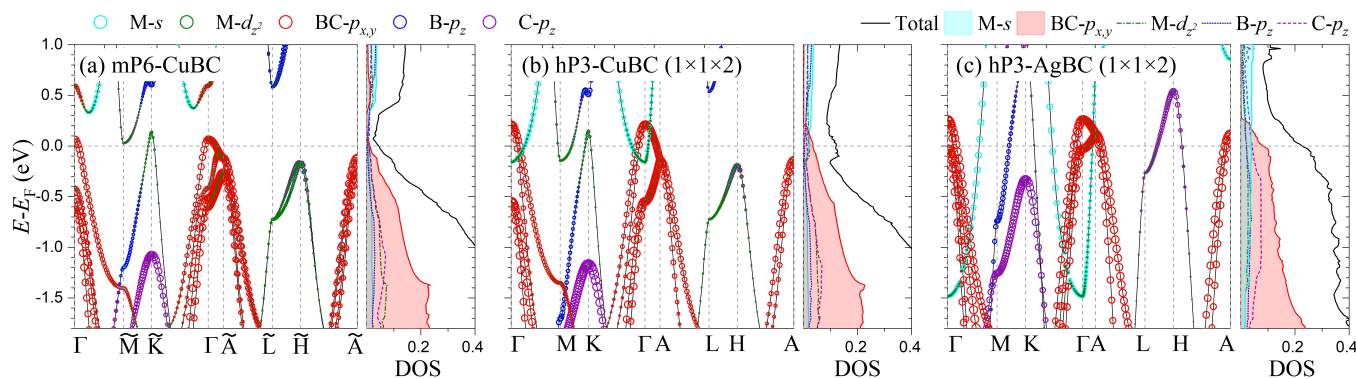


FIG. 6. Orbital-resolved electronic band structure and DOS in states/(eV atom) in CuBC and AgBC phases with straight (hP3) and tilted (mP6) C-M-C dumbbells. For direct comparison, the hP3 unit cells are doubled along the c axis, and the high-symmetry points along the standard path for mP6 are labeled using hexagonal notation ($\tilde{M}=Z$, $\tilde{K}=(1/3, 1/3, 0)$, $\tilde{A}=Y$, $\tilde{L}=D$, $\tilde{H}=(1/3, 1/3, 1/2)$). The $p_{x,y}$ characters of B and C are combined, and only the d_{z^2} character of the d manifold is shown for Cu or Ag, since the combined contribution from all other d -states is five times smaller at the Fermi level. The key feature defining the hole-doping level of the BC- $p_{x,y}$ states is the position of the M- s band edge: (a) above E_F ; (b) -0.15 eV below E_F ; and (c) -1.5 eV below E_F .

for $y = 1$ and $y = 1/2$ indicate that more stable hP12 ($P3m1$) and hP30 configurations with metal blocks separated from bridged BC frameworks exist even in the stoichiometric CuBC and AgBC. To illustrate that this is not an uncommon situation, we performed evolutionary searches for AgGaO₂ and found an hP12 ($R\bar{3}m$) structure with an alternative morphology 0.12 eV/atom below the synthesized wurtzite prototype (see Supplementary Fig. 4). Therefore, one can expect that no major rebonding will occur in the topochemical Li-to-Ag ion exchange for borocarbides either.

To identify possible products of partial ion exchange in stoichiometric and delithiated Li _{x} BC precursors, we constructed supercells with $x = 1, 7/8, 5/6, 3/4, 2/3, 5/8$, and $1/2$ and screened over 3600 layered Li _{$x-y$} M _{y} BC configurations for each transition metal. Representative relative stabilities of quaternary compounds are shown in Fig. 4 for $x = 2/3$, with results for other compositions compiled in Supplementary Fig. 5 and the locally stable phases at 600 K collected in Fig. 5.

The results show that ordered mixed-metal derivatives with Cu and Ag below the Li _{x} BC and M _{x} BC line are present at each starting composition x . The two structures displayed in Fig. 4 reflect the observed preferences for Cu to co-exist with Li within galleries and for Ag to avoid mixing. 13 Cu-based and 7 Ag-based quaternary compounds are locally stable, by ΔE_{rel} between -0.004 and -0.040 eV/atom at $T = 0$ K. The vibrational entropy contribution at 600 K moves the reaction free energies upward by an average of 0.014 eV/atom and disfavors some of the quaternary phases by up to 0.011 eV/atom, fully destabilizing two of them. For comparison, the configuration entropy stabilization estimated at $T = 600$ K amounts to only -0.012 eV/atom (see Supplementary Note II). These contributions are considerably smaller in magnitude than the reaction energies of full

ion exchange of about -0.4 eV/atom. In light of the previously predicted and observed transformations [30], the topochemical reactions are expected to proceed until full Li to Cu or Ag ion exchange. If quaternary compounds do form, they may have disordered double-metal arrangements, unless the samples can be annealed.

The Li occupancy pattern in partially delithiated Li _{x} BC precursors may also play a critical role in defining the reaction products. While staging models were proposed to explain lithium-deficient borocarbides [39], the lowest-energy configurations in our DFT calculations at all compositions featured uniform distributions of Li along the c axis [40]. Therefore, the most stable identified Li _{$x-y$} M _{y} BC derivatives with uneven metal content per layer, such as hP32-Li_{1/6}Cu_{1/2}BC with fully Ag-filled and metal-free galleries (Fig. 4) as part of the most viable series with $y = 1/2$ for both transition metals (Fig. 5), might not be reachable kinetically. Nevertheless, the representative models with relatively small unit cells are useful for investigating various properties of the mixed-metal borocarbides. In particular, the 60 meV/Å² binding energy per area between BC-Ag-BC-Li_{2/3}-BC-Ag-BC blocks makes the hP32 quaternary phase potentially exfoliable [80, 81].

C. Electronic and superconducting properties

Stoichiometric layered borocarbides of alkali and alkaline-earth metals exhibit semiconducting behavior expected for materials satisfying the 8-electron rule. Early studies with standard semi-local functionals found similar band gaps of ~ 1 eV in LiBC [32], MgB₂C₂ [82], and BeB₂C₂ [42]. The band gaps in our HSE06 (optB86b-vdW) calculations are 1.61 (0.84) eV in LiBC, 1.98 (1.18) eV in MgB₂C₂, 1.29 (0.39) eV in BeB₂C₂,



TABLE II. Projected BC- $p_{x,y}$ density of states at the Fermi level, $N_{p_{x,y}}(E_F)$ (states/(eV atom)), logarithmic average phonon frequency, $\omega_{\log}$ (meV), and superconducting critical temperature, T_c , calculated using the Allen-Dynes formula [83] with $\mu^* = 0.1$, the iME formalism with $\mu^* = 0.1$ (e-ph matrix elements computed with QUANTUM ESPRESSO on coarse grids), or the aME formalism with $\mu^* = 0.2$ (e-ph matrix elements computed with EPW on fine grids) for select metal borocarbides. For comparison, we list MgB₂ properties evaluated with similar settings in our previous study [64]. Among the considered quaternary phases, only Li_{1/2}Ag_{1/4}BC is above the local convex hull (by 12 meV/atom).

Composition	Pearson symbol	$N_{p_{x,y}}$ at E_F	$\omega_{\log}$ (meV)	AD T_c (K)	iME T_c (K)	aME T_c (K)
MgB ₂	hP3	0.293	60.3	18.1	24.9	41
CuBC	mP6	0.009	19.9	0.3	0.4	
Cu _{2/3} BC	hP16	0.097	34.9	14.7	16.0	
Cu _{5/8} BC	hP21	0.119	42.8	23.4	26.6	
AgBC	hP3	0.083	50.5	10.6	12.0	56
Ag _{1/2} BC	oP10	0.069	20.1	7.0	6.5	
Li _{1/6} Cu _{2/3} BC	hP17	0.055	35.5	2.2	2.5	
Li _{1/2} Cu _{1/6} BC	hP16	0.108	63.0	11.3	12.9	
Li _{1/2} Ag _{1/4} BC	hP22	0.095	57.7	10.9	12.7	
Li _{1/4} Ag _{1/2} BC	oI22	0.099	55.5	15.7	18.2	

1.87 (0.95) eV in NaBC, and 1.67 (0.95) eV in ZnB₂C₂. The findings underscore yet again the uniqueness of the isoelectronic but intrinsically hole-doped MgB₂ thermodynamically stable under ambient conditions.

Replacing Li with Cu or Ag turns the stoichiometric borocarbides into metals and potential MgB₂-type superconductors. To disentangle the dependence of the electronic states on the structural and chemical factors, we display projections of the most relevant states around the Fermi level for three layered phases in Fig. 6. Due to the strong mixing of B and C in-plane orbitals, we group their contributions as BC- $p_{x,y}$ character. The interlayer bridging by the metal-carbon dumbbells makes the M- d_{z^2} states hybridize with the C- p_z , B- p_z , and M- s states near the Fermi level in CuBC. A key feature in both hP3 phases is the partial occupation of the nearly-free electron M- s band, which contributes significantly to the total DOS at the Fermi level and results in hole doping of the BC- $p_{x,y}$ states. In the mP6-CuBC derivative with only 0.5 % shorter interlayer spacing, the M- s states are no longer filled and the in-plane BC states are barely hole doped, which leads to the emergence of a pseudogap. The much deeper position of the M- s band edge in hP3-AgBC, at -1.5 eV at Γ , prevents the material from emptying the nearly free electron states and reducing the DOS at the Fermi level via a similar structural distortion. Additional tests in Supplementary Fig. 6 reveal that the size difference between the two transition metals, that determines the interlayer distance, is the primary factor for the substantially different electronic landscapes in the

two layered stoichiometric materials. While AgBC is dynamically stable at 0 GPa, both CuBC and AgBC are (further) destabilized under pressure (see Supplementary Fig. 7).

The projected DOS map in Fig. 5 shows that AgBC is the only viable considered compound with substantially hole-doped BC- $p_{x,y}$ states in the fully intercalated Li_{1-y}M_yBC series. In order to achieve similar doping levels of the covalent framework in non-stoichiometric borocarbides, one has to start with sufficiently delithiated Li_xBC precursors. Namely, the first locally stable phases with at least 0.08 states/(eV atom) BC- $p_{x,y}$ contribution appear at $x = 7/8$ and $x = 3/4$ in the Cu- and Ag-based quaternaries, respectively. The low projected DOS values in some mixed-metal compounds with depleted metal content, e.g., in Li_{1/3}Cu_{1/3}BC, indicate a complex interplay between competing states at the Fermi level. In fact, metal filling reduction appears necessary for the onset of superconductivity in Cu_yBC ($y = 2/3$) but would be detrimental for the T_c in Ag_yBC.

The following superconductivity analysis, based on isotropic Migdal-Eliashberg (iME) estimates of T_c , supports these qualitative observations. The results listed in Table II and Supplementary Fig. 8 show that CuBC has negligible T_c at $y = 1$ but becomes a promising superconductor at lower metal concentrations, while AgBC has a higher T_c estimate than the half-full counterpart. The phonon dispersions in Supplementary Fig. 9 show that minor dynamical instabilities, common in layered materials [84], persist in some of the identified phases and would need to be carefully addressed, but the presented Eliashberg spectral functions indicate that these materials should have strong e-ph coupling [85]. Overall, several ternary and quaternary borocarbides with the transition metals have T_c between 10 and 23 K, comparable to the values in Mg-Li-B [64], Li-M-BC [40], and Na(Mg)-BC [43] obtained at the same level of theory. Since these studies have shown a general underestimation of the isotropic T_c by a factor of two to four in such layered materials, we have performed aME calculations for the representative AgBC material.

The results in Fig. 7 establish hP3-AgBC as a compelling superconductor that merges key features of MgB₂ [86, 87] and graphite intercalation compounds (GICs) [88, 89], yet deviates from their prototypical behavior. The characteristic hourglass-shaped Fermi surfaces from heavily hole-doped B- $p_{x,y}$ states in MgB₂ [64, 87] or essentially cylindrical sheets in LiB [64, 90, 91] are replaced by elongated ellipsoids in AgBC, a consequence of an insufficient doping of the BC- $p_{x,y}$ bands that now cross the Fermi level between Γ and A. The BC- p_z bands generate closed sheets around the H point with the triangular cross-section in the A-H-L plane. In contrast to the electronically inert Mg²⁺, Ag retains some charge on its s orbitals that generate Fermi surfaces enclosing the Γ point, but the pronounced c -axis anisotropy deforms the NFE-type spheres observed in CaC₆ [92] into pancake shapes.



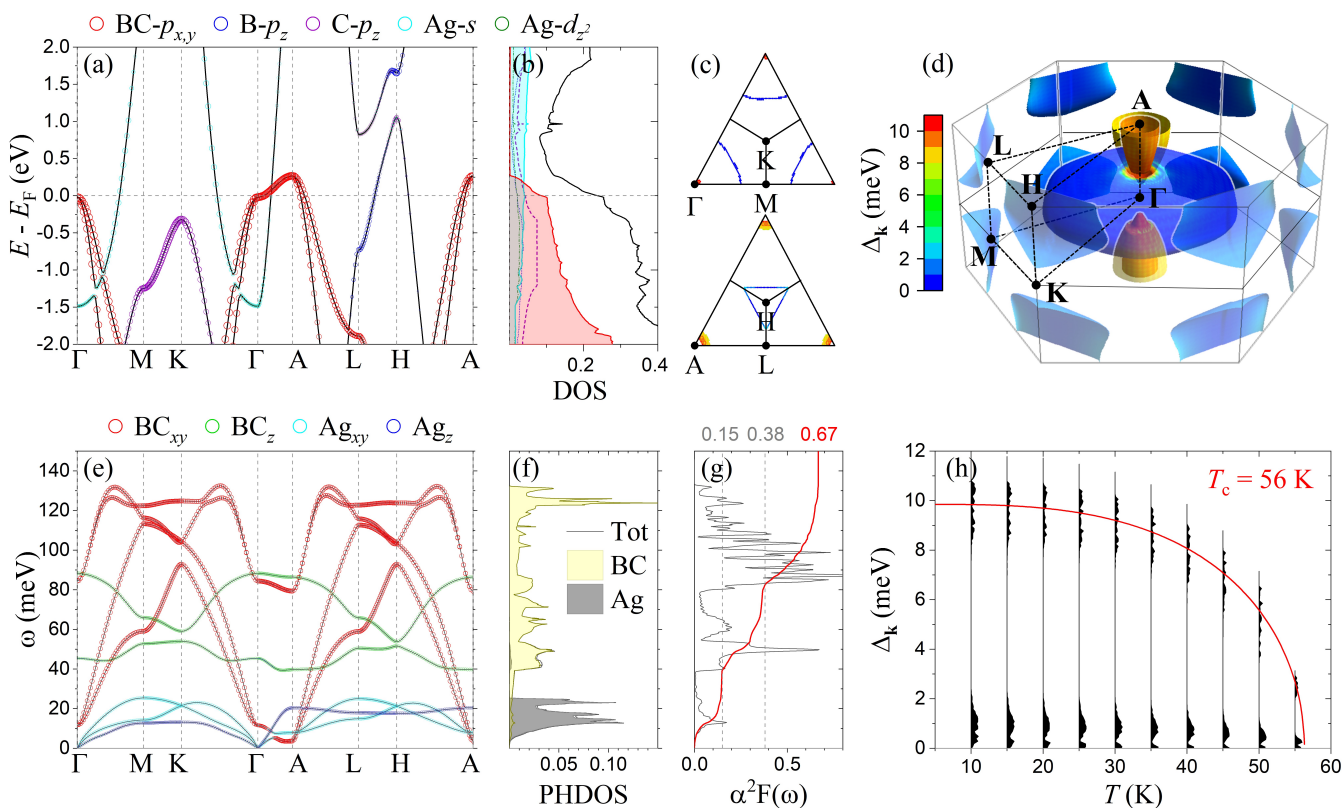


FIG. 7. Electronic, vibrational, and superconducting properties of hP3-AgBC. (a) Orbital-resolved band structure. (b) Total and projected DOS in states/(eV atom). (c) Fermi surface cuts in the Γ -M-K and A-L-H planes. (d) Fermi surfaces with the color map showing superconducting gap values. (e) Phonon dispersion with atom- and direction-resolved eigenmodes. (f) Phonon DOS in states/(meV atom) projected on BC and Ag. (g) Eliashberg spectral function and e-ph coupling strength integrated to 0.67. (h) Superconducting gap as a function of temperature extrapolated to vanish at $T_c = 56$ K for $\mu^* = 0.2$.

The existence of the three large Fermi surface pockets defines three sizable contributions to the e-ph coupling (λ) that can be distinguished by examining the phonon dispersion and Eliashberg spectral function displayed in Fig. 7(e-g). In contrast to MgB₂ with the same $\lambda = 0.67$ [64], only about 0.30 (40%) arises from the in-plane bond stretching mode E' coupled with the BC- $p_{x,y}$ states, manifested through the pronounced ~ 30 % phonon softening down to 85 meV at Γ and the familiar splitting of these electronic states under the frozen phonon displacements (Supplementary Fig. 10). The rest comes from BC_z phonons in the 40-60 meV range, particularly along the M-K and L-H directions, and soft mixed modes around 10-20 meV as further evidenced by the mode and $\mathbf{q}$ -point resolved λ (see Supplementary Fig. 11). The in-plane intercalant phonons play a dominant role in CaC₆ [88] but not in AgBC, where Ag is linked with C in dumbbells.

To estimate the superconducting T_c , we solved the anisotropic Migdal-Eliashberg (aME) equations and tracked the evolution of the superconducting gap $\Delta_{\mathbf{k}}$ with temperature. As shown in Fig. 7(h), AgBC exhibits two-gap superconductivity with pronounced anisotropy, similar to MgB₂. The smaller gap lies below 2 meV as-

sociated with the pancake-shaped Fermi surface around Γ and the closed pocket surrounding H, while the larger has an average around 10 meV, residing on the ellipsoidal surfaces linked to the BC- $p_{x,y}$ states. Both gaps close around $T = 56$ K for a Coulomb pseudopotential $\mu^* = 0.2$, chosen such that the experimental T_c of MgB₂ is reproduced under similar computational settings [64]. The comparison of logarithmic average phonon frequency and the corresponding Allen-Dynes estimates listed in Table II illustrates that higher $\omega_{\log}$ values (60.3 meV in MgB₂ versus 50.5 meV in AgBC) do not necessarily mean a higher T_c in anisotropic materials with multiple gaps. Due to the nearly five-fold boost in T_c , we expect the other iME critical temperature estimates to be increased substantially as well. Lastly, we investigated the effect of doping AgBC by also calculating the aME T_c after rigidly shifting the Fermi level by ± 0.1 eV. The results in Supplementary Fig. DG 12 show that moderate electron-doping decreases the T_c down to 36 K, stemming from the loss of BC- $p_{x,y}$ states, while hole-doping increases the T_c to 61 K.



IV. Discussion and Conclusions

The presented findings provide further evidence that borocarbides comprising honeycomb layers intercalated with low-valent metals hold great promise as high- T_c and potentially practical superconductors [40, 43]. Such compounds are uniquely suited to possess hole-doped covalent bonds requisite for MgB_2 -type superconductivity. Indeed, the B-B honeycomb framework is intrinsically two electrons deficient and, once stabilized, resists chemical modifications that further hole-dope the $\text{B-}p_{x,y}$ states [4, 93]. Graphite is already stable and easy to intercalate, but the resulting superconductors with electron-doped NFE states exhibit relatively low T_c [88, 94], with potential to exceed 30 K, e.g., in Na-C, under pressure [84, 95]. While the B-C framework represents a favorable middle ground between the chemically active B-B and the relatively inert C-C counterparts, the choice of the intercalant remains important: the larger and higher-valent metals like Ca and Sr tend to stabilize clathrate or layered but non-honeycomb structures that lack the key MgB_2 features [96, 97].

The proposed topochemical ion exchange offers some practical advantages over our previously considered soft chemistry reintercalation of layered metal borocarbides [40, 43, 44]. The favorable thermodynamics of the exchange reaction is ensured not through initial delithiation of LiBC but rather through the introduction of new metals in the form of halide or nitrate salts, which helps avoid the high-temperature deintercalation process and reduce the possibility of structural disorder in the precursor. Replacement of Li with Cu or Ag, successfully achieved in wurtzite-type oxides [30] with arguably less well-defined layered frameworks shown to be metastable in our study, can be expected to be more natural in the case of honeycomb LiBC. Synthesis of the metastable CuBC or AgBC identified with extensive structure searches would represent the first *theory-guided* topochemical ion exchange producing closely related but *non-isostructural* derivatives. The existence of Li_xBC

precursors greatly expands the space of viable products to non-stoichiometric ternary or quaternary derivatives and can aid in the rational design of superconductors.

Remarkably, the accessible stoichiometric AgBC with the simplest considered three-atom unit cell already embodies the long-sought features for conventional superconductivity. Compared to the MgB_2 , CaC_6 , and predicted LiB superconductors, AgBC benefits from all three e-ph coupling channels present in honeycomb compounds. According to our aME calculations, the more balanced contributions result in T_c values reaching 56 K. Modest hole-doping could further boost T_c by turning ellipsoidal $\text{BC-}p_{x,y}$ Fermi sheets into hallmark 2D cylinders. CuBC is predicted to be a frustrated metal with low DOS at the Fermi level, but topochemical synthesis from Li_xBC may produce even better Cu_xBC superconductors. Altogether, the reintercalation and topochemical pathways proposed in our previous [40, 43, 44] and present studies, respectively, establish layered metal borocarbides as primary candidates to superconduct at near or above liquid nitrogen temperature.

Acknowledgments

We thank Igor Mazin for valuable discussions and Christopher Renskers for his help in identifying the CuBC structural model. The authors acknowledge support from the National Science Foundation (NSF) (Award No. DMR-2132589). This work used the Expanse system at the San Diego Supercomputer Center via allocation TG-DMR180071 and the Frontera supercomputer at the Texas Advanced Computing Center via the Leadership Resource Allocation (LRAC) award DMR22004. Expanse is supported by the Extreme Science and Engineering Discovery Environment (XSEDE) program [98] through NSF Award No. ACI-1548562, and Frontera is supported by NSF Award No. OAC-1818253 [99].

- [1] J. A. Flores-Livas, L. Boeri, A. Sanna, G. Profeta, R. Arita, and M. Eremets, A perspective on conventional high-temperature superconductors at high pressure: Methods and materials, *Physics Reports* **856**, 1 (2020).
- [2] L. Boeri, R. Hennig, P. Hirschfeld, G. Profeta, A. Sanna, E. Zurek, W. E. Pickett, M. Amsler, R. Dias, M. I. Eremets, C. Heil, R. J. Hemley, H. Liu, Y. Ma, C. Pierleoni, A. N. Kolmogorov, N. Rybin, D. Novoselov, V. Anisimov, A. R. Oganov, C. J. Pickard, T. Bi, R. Arita, I. Errea, C. Pellegrini, R. Requist, E. K. U. Gross, E. R. Margine, S. R. Xie, Y. Quan, A. Hire, L. Fanfarillo, G. R. Stewart, J. J. Hamlin, V. Staney, R. S. Gonnelli, E. Piatti, D. Romanin, D. Daghero, and R. Valenti, The 2021 room-temperature superconductivity roadmap, *Journal of Physics: Condensed Matter* **34**, 183002 (2022).
- [3] L. R. Testardi, Structural instability and superconductivity in A-15 compounds, *Reviews of Modern Physics* **47**, 637–648 (1975).
- [4] A. N. Kolmogorov, M. Calandra, and S. Curtarolo, Thermodynamic stabilities of ternary metal borides: An ab initio guide for synthesizing layered superconductors, *Phys. Rev. B* **78**, 094520 (2008).
- [5] S. Shah and A. N. Kolmogorov, Stability and superconductivity of Ca-B phases at ambient and high pressure, *Phys. Rev. B* **88**, 014107 (2013).
- [6] T. F. T. Cerqueira, Y.-W. Fang, I. Errea, A. Sanna, and M. A. L. Marques, Searching Materials Space for Hydride Superconductors at Ambient Pressure, *Advanced Functional Materials* **34**, 2404043 (2024).
- [7] R. Wang, F. Zheng, Z. Zhang, S. Wu, H. Dong, C.-Z.



- Wang, V. Antropov, Y. Sun, and K.-M. Ho, Anticorrelation between electron-phonon coupling strength and stability of ternary metal diborides, *Physical Review B* **111**, 10.1103/physrevb.111.014104 (2025).
- [8] A. P. Drozdov, M. I. Erements, I. A. Troyan, V. Ksenofontov, and S. I. Shylin, Conventional superconductivity at 203 kelvin at high pressures in the sulfur hydride system, *Nature* **525**, 73 (2015).
- [9] A. P. Drozdov, P. P. Kong, V. S. Minkov, S. P. Besedin, M. A. Kuzovnikov, S. Mozaffari, L. Balicas, F. F. Balakirev, D. E. Graf, V. B. Prakapenka, E. Greenberg, D. A. Knyazev, M. Tkacz, and M. I. Erements, Superconductivity at 250 K in lanthanum hydride under high pressures, *Nature* **569**, 528 (2019).
- [10] E. Hertel and K. Schneider, Umwandlungen im Kristallgitter, *Zeitschrift für Elektrochemie und angewandte physikalische Chemie* **37**, 536 (1931).
- [11] M. D. Cohen and G. M. J. Schmidt, 383. Topochemistry. Part I. A survey, *Journal of the Chemical Society*, 1996 (1964).
- [12] Y. Hosogi, H. Kato, and A. Kudo, Visible light response of $\text{AgLi}_{1/3}\text{M}_{2/3}\text{O}_2$ ($\text{M} = \text{Ti}$ and Sn) synthesized from layered Li_2MO_3 using molten AgNO_3 , *Journal of Materials Chemistry C* **18**, 647 (2008).
- [13] H. Gwon, S.-W. Kim, Y.-U. Park, J. Hong, G. Ceder, S. Jeon, and K. Kang, Correction to Ion-Exchange Mechanism of Layered Transition-Metal Oxides: Case Study of $\text{LiNi}_{0.5}\text{Mn}_{0.5}\text{O}_2$, *Inorganic Chemistry* **54**, 2081 (2015).
- [14] J. H. Roudebush, K. A. Ross, and R. J. Cava, Iridium containing honeycomb delafossites by topotactic cation exchange, *Dalton Transactions* **45**, 8783 (2016).
- [15] M. Abramchuk, C. Ozsoy-Keskinbora, J. W. Krizan, K. R. Metz, D. C. Bell, and F. Tafti, Cu_2IrO_3 : A New Magnetically Frustrated Honeycomb Iridate, *Journal of the American Chemical Society* **139**, 15371 (2017).
- [16] A. Sethi, D. K. Yadav, and S. Uma, Cuprous delafossites, $\text{Cu}_3(\text{MFeSb})\text{O}_6$ ($\text{M} = \text{Na}, \text{Li}$) with honeycomb arrays, realized by topochemical ion-exchange reactions, *Ceramics International* **48**, 13833 (2022).
- [17] H. Boller and K. Hiebl, Quaternary pseudo-intercalation phases $\text{T}_x[\text{Nb}_2\text{S}_2\text{C}]$ ($\text{T} = \text{V}, \text{Cr}, \text{Mn}, \text{Fe}, \text{Co}, \text{Ni}, \text{Cu}$) and metastable $\text{Nb}_2\text{S}_2\text{C}$ formed by topochemical synthesis, *Journal of Alloys and Compounds* **183**, 438 (1992).
- [18] O. J. Rutt, G. R. Williams, and S. J. Clarke, Reversible lithium insertion and copper extrusion in layered oxysulfides, *Chemical Communications*, 2869 (2006).
- [19] A. J. Martinolich and J. R. Neilson, Toward reaction-by-design: Achieving kinetic control of solid state chemistry with metathesis, *Chemistry of Materials* **29**, 479 (2017).
- [20] S. Sasaki, S. Giri, S. J. Cassidy, S. Dey, M. Batuk, D. Vandemeulebroucke, G. Cibir, R. I. Smith, P. Holdship, C. P. Grey, J. Hadermann, and S. J. Clarke, Anion redox as a means to derive layered manganese oxychalcogenides with exotic intergrowth structures, *Nature Communications* **14**, 2917 (2023).
- [21] G. Bhaskar, V. Gvozdetzkyi, M. Batuk, K. M. Wiaderek, Y. Sun, R. Wang, C. Zhang, S. L. Carnahan, X. Wu, R. A. Ribeiro, S. L. Bud'ko, P. C. Canfield, W. Huang, A. J. Rossini, C.-Z. Wang, K.-M. Ho, J. Hadermann, and J. V. Zaikina, Topochemical Deintercalation of Li from Layered LiNiB : Toward 2D MBene, *Journal of the American Chemical Society* **143**, 4213 (2021).
- [22] K. J. Baumler, O. S. Adams, and R. E. Schaak, One-step topochemical transformation of MoAlB into metastable Mo_2AlB_2 using a metal chloride salt reaction, *Chemical Communications* **59**, 4814 (2023).
- [23] J. Mou, W. Zhang, S. Li, X. Fan, and X. Zhang, Topochemical synthesis of Mo_2AlB_2 with enhanced electrochemical properties, *Journal of the European Ceramic Society* **45**, 117184 (2025).
- [24] F. Bahrami, W. Lafargue-Dit-Hauret, O. I. Lebedev, R. Movshovich, H.-Y. Yang, D. Broido, X. Rocquefelte, and F. Tafti, Thermodynamic Evidence of Proximity to a Kitaev Spin Liquid in $\text{Ag}_3\text{LiIr}_2\text{O}_6$, *Physical Review Letters* **123**, 237203 (2019).
- [25] D. Li, K. Lee, B. Y. Wang, M. Osada, S. Crossley, H. R. Lee, Y. Cui, Y. Hikita, and H. Y. Hwang, Superconductivity in an infinite-layer nickelate, *Nature* **572**, 624–627 (2019).
- [26] S. Bette, T. Takayama, V. Duppel, A. Poulain, H. Takagi, and R. E. Dinnebier, Crystal structure and stacking faults in the layered honeycomb, delafossite-type materials $\text{Ag}_3\text{LiIr}_2\text{O}_6$ and $\text{Ag}_3\text{LiRu}_2\text{O}_6$, *Dalton Transactions* **48**, 9250 (2019).
- [27] L. T. Alameda, R. W. Lord, J. A. Barr, P. Moradifar, Z. P. Metzger, B. C. Steimle, C. F. Holder, N. Alem, S. B. Sinnott, and R. E. Schaak, Multi-Step Topochemical Pathway to Metastable Mo_2AlB_2 and Related Two-Dimensional Nanosheet Heterostructures, *Journal of the American Chemical Society* **141**, 10852 (2019).
- [28] L. Si, W. Xiao, J. Kaufmann, J. M. Tomczak, Y. Lu, Z. Zhong, and K. Held, Topotactic hydrogen in nickelate superconductors and akin infinite-layer oxides abO_2 , *Phys. Rev. Lett.* **124**, 166402 (2020).
- [29] J. V. Handy, J. L. Andrews, B. Zhang, D. Kim, N. Bhuvanesh, Q. Tu, X. Qian, and S. Banerjee, Topochemical stabilization and single-crystal transformations of a metastable 2D $\gamma\text{-V}_2\text{O}_5$ intercalation cathode, *Cell Reports Physical Science* **3**, 100712 (2022).
- [30] I. Suzuki, M. Kita, and T. Omata, Designing Topotactic Ion-Exchange Reactions in Solid-State Oxides Through First-Principles Calculations, *Chemistry of Materials* **36**, 4196 (2024).
- [31] M. Wörle, R. Nesper, G. Mair, M. Schwarz, and H. G. V. Schnering, LiBC — ein vollständig interkalierter Heterographit, *ZAAC - Journal of Inorganic and General Chemistry* **621**, 1153 (1995).
- [32] H. Rosner, A. Kitaigorodsky, and W. E. Pickett, Prediction of high T_c superconductivity in hole-doped LiBC , *Physical Review Letters* **88**, 127001 (2002).
- [33] J. K. Dewhurst, S. Sharma, C. Ambrosch-Draxl, and B. Johansson, First-principles calculation of superconductivity in hole-doped LiBC : T_c 65 K, *Physical Review B* **68**, 020504 (2003).
- [34] A. Bharathi, S. J. Balaseli, M. Premila, T. Sairam, G. Reddy, C. Sundar, and Y. Hariharan, Synthesis and search for superconductivity in LiBC , *Solid State Communications* **124**, 423 (2002).
- [35] L. Zhao, P. Klavins, and K. Liu, Synthesis and properties of hole-doped Li_{1-x}BC , *Journal of Applied Physics* **93**, 8653 (2003).
- [36] A. M. Fogg, J. B. Claridge, G. R. Darling, and M. J. Rosseinsky, Synthesis and characterization of Li_xBC —hole doping does not induce superconductivity, *Chemical Communications* **3**, 1348 (2003).
- [37] A. M. Fogg, P. R. Chalker, J. B. Claridge, G. R. Darling, and M. J. Rosseinsky, LiBC electronic, vibrational, structural, and low-temperature chemical behavior of a



- layered material isoelectronic with MgB_2 , *Physical Review B* **67**, 245106 (2003).
- [38] A. M. Fogg, J. Meldrum, G. R. Darling, J. B. Claridge, and M. J. Rosseinsky, Chemical Control of Electronic Structure and Superconductivity in Layered Borides and Borocarbides: Understanding the Absence of Superconductivity in Li_xBC , *Journal of the American Chemical Society* **128**, 10043 (2006).
- [39] B. Kalkan and E. Ozdas, Staging phenomena in lithium-intercalated boron-carbon, *ACS Applied Materials & Interfaces* **11**, 4111 (2019).
- [40] C. R. Tomassetti, G. P. Kafle, E. T. Marcial, E. R. Margine, and A. N. Kolmogorov, Prospect of high-temperature superconductivity in layered metal borocarbides, *Journal of Materials Chemistry C* **12**, 4870 (2024).
- [41] M. Wörle and R. Nesper, MgB_2C_2 , a new graphite-related refractory compound, *Journal of Alloys and Compounds* **216**, 75 (1994).
- [42] K. Hofmann, X. Rocquefelte, J. F. Halet, C. Bähitz, and B. Albert, The η_6, η_1 -coordination of beryllium atoms in the graphite analogue BeB_2C_2 , *Angewandte Chemie - International Edition* **47**, 2301 (2008).
- [43] C. R. Tomassetti, D. Gochitashvili, C. Renskers, E. R. Margine, and A. N. Kolmogorov, First-principles design of ambient-pressure $\text{Mg}_x\text{B}_2\text{C}_2$ and Na_xBC superconductors, *Physical Review Materials* **8**, 114801 (2024).
- [44] S. Kharabade, M. Meyers, C. R. Tomassetti, E. R. Margine, I. I. Mazin, and A. N. Kolmogorov, Thermodynamic stability of Li-B-C compounds from first principles, *Physical Chemistry Chemical Physics* **25**, 7344 (2023).
- [45] E. R. Margine and F. Giustino, Anisotropic Migdal-Eliashberg theory using Wannier functions, *Physical Review B* **87**, 024505 (2013).
- [46] R. Lucrezi, P. P. Ferreira, M. Aichhorn, and C. Heil, Temperature and quantum anharmonic lattice effects on stability and superconductivity in lutetium trihydride, *Nature Communications* **15**, 441 (2024).
- [47] G. Kresse and J. Furthmüller, Efficient iterative schemes for *ab initio* total-energy calculations using a plane-wave basis set, *Physical Review B* **54**, 11169 (1996).
- [48] P. E. Blöchl, Projector augmented-wave method, *Physical Review B* **50**, 17953 (1994).
- [49] A. N. Kolmogorov and V. H. Crespi, Registry-dependent interlayer potential for graphitic systems, *Physical Review B* **71**, 235415 (2005).
- [50] S. Lebègue, J. Harl, T. Gould, J. G. Ángyán, G. Kresse, and J. F. Dobson, Cohesive properties and asymptotics of the dispersion interaction in graphite by the random phase approximation, *Physical Review Letters* **105**, 196401 (2010).
- [51] A. N. Kolmogorov, S. Hajinazar, C. Angyal, V. L. Kuznetsov, and A. P. Jephcoat, Synthesis of a predicted layered LiB via cold compression, *Physical Review B* **92**, 144110 (2015).
- [52] J. Ning, M. Kothakonda, J. W. Furness, A. D. Kaplan, S. Ehlert, J. G. Brandenburg, J. P. Perdew, and J. Sun, Workhorse minimally empirical dispersion-corrected density functional with tests for weakly bound systems: $\text{r}^2\text{SCAN}+\text{rVV10}$, *Physical Review B* **106**, 075422 (2022).
- [53] J. Klimeš, D. R. Bowler, and A. Michaelides, Van der Waals density functionals applied to solids, *Physical Review B* **83**, 195131 (2011).
- [54] J. W. Furness, A. D. Kaplan, J. Ning, J. P. Perdew, and J. Sun, Accurate and Numerically Efficient r^2SCAN Meta-Generalized Gradient Approximation, *The Journal of Physical Chemistry Letters* **11**, 8208 (2020).
- [55] A. V. Krukau, O. A. Vydrov, A. F. Izmaylov, and G. E. Scuseria, Influence of the exchange screening parameter on the performance of screened hybrid functionals, *The Journal of Chemical Physics* **125**, 224106 (2006).
- [56] H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Physical Review B* **13**, 5188 (1976).
- [57] S. Hajinazar, A. Thorn, E. D. Sandoval, S. Kharabade, and A. N. Kolmogorov, MAISE: Construction of neural network interatomic models and evolutionary structure optimization, *Computer Physics Communications* **259**, 107679 (2021).
- [58] A. N. Kolmogorov, S. Shah, E. R. Margine, A. F. Bialon, T. Hammerschmidt, and R. Drautz, New Superconducting and Semiconducting Fe-B Compounds Predicted with an *Ab Initio* Evolutionary Search, *Physical Review Letters* **105**, 217003 (2010).
- [59] A. Togo and I. Tanaka, First principles phonon calculations in materials science, *Scripta Materialia* **108**, 1 (2015).
- [60] P. Giannozzi, O. Andreussi, T. Brumme, O. Bunau, M. Buongiorno Nardelli, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, M. Cococcioni, N. Colonna, I. Carnimeo, A. Dal Corso, S. de Gironcoli, P. Delugas, R. A. DiStasio, A. Ferretti, A. Floris, G. Fratesi, G. Fugallo, R. Gebauer, U. Gerstmann, F. Giustino, T. Gorni, J. Jia, M. Kawamura, H.-Y. Ko, A. Kokalj, E. Küçükbenli, M. Lazzeri, M. Marsili, N. Marzari, F. Mauri, N. L. Nguyen, H.-V. Nguyen, A. Otero-de-la Roza, L. Paulatto, S. Poncè, D. Rocca, R. Sabatini, B. Santra, M. Schlipf, A. P. Seitsonen, A. Smogunov, I. Timrov, T. Thonhauser, P. Umari, N. Vast, X. Wu, and S. Baroni, Advanced capabilities for materials modelling with Quantum ESPRESSO, *Journal of Physics: Condensed Matter* **29**, 465901 (2017).
- [61] M. J. van Setten, M. Giantomassi, E. Bousquet, M. J. Verstraete, D. R. Hamann, X. Gonze, and G.-M. Rignanese, The PseudoDojo: Training and grading a 85 element optimized norm-conserving pseudopotential table, *Computer Physics Communications* **226**, 39 (2018).
- [62] M. Methfessel and A. T. Paxton, High-precision sampling for Brillouin-zone integration in metals, *Physical Review B* **40**, 3616 (1989).
- [63] S. Baroni, S. De Gironcoli, A. Dal Corso, and P. Giannozzi, Phonons and related crystal properties from density-functional perturbation theory, *Reviews of Modern Physics* **73**, 515 (2001).
- [64] G. P. Kafle, C. R. Tomassetti, I. I. Mazin, A. N. Kolmogorov, and E. R. Margine, *Ab initio* study of Li-Mg-B superconductors, *Physical Review Materials* **6**, 084801 (2022).
- [65] F. Giustino, M. L. Cohen, and S. G. Louie, Electron-phonon interaction using Wannier functions, *Physical Review B* **76**, 165108 (2007).
- [66] S. Poncè, E. R. Margine, C. Verdi, and F. Giustino, EPW: Electron-phonon coupling, transport and superconducting properties using maximally localized Wannier functions, *Computer Physics Communications* **209**, 116 (2016).
- [67] H. Lee, S. Poncè, K. Bushick, S. Hajinazar, J. Lafuente-



- Bartolome, J. Leveillee, C. Lian, J.-M. Lihm, F. Macheda, H. Mori, H. Paudyal, W. H. Sio, S. Tiwar, M. Zacharias, X. Zhang, N. Bonini, E. Kioupakis, E. R. Margine, and F. Giustino, Electron-phonon physics from first principles using the EPW code, *npj Computational Materials* **9**, 156 (2023).
- [68] N. Marzari, A. A. Mostofi, J. R. Yates, I. Souza, and D. Vanderbilt, Maximally localized Wannier functions: Theory and applications, *Reviews of Modern Physics* **84**, 1419 (2012).
- [69] G. Pizzi, V. Vitale, R. Arita, S. Blügel, F. Freimuth, G. Géranton, M. Gibertini, D. Gresch, C. Johnson, T. Koretsune, J. Ibañez-Azpiroz, H. Lee, J.-M. Lihm, D. Marchand, A. Marrazzo, Y. Mokrousov, J. I. Mustafa, Y. Nohara, Y. Nomura, L. Paulatto, S. Poncé, T. Ponweiser, J. Qiao, F. Thöle, S. S. Tsirkin, M. Wierzbowska, N. Marzari, D. Vanderbilt, I. Souza, A. A. Mostofi, and J. R. Yates, Wannier90 as a community code: new features and applications, *Journal of Physics: Condensed Matter* **32**, 165902 (2020).
- [70] A. Marrazzo, S. Beck, E. R. Margine, N. Marzari, A. A. Mostofi, J. Qiao, I. Souza, S. S. Tsirkin, J. R. Yates, and G. Pizzi, Wannier-function software ecosystem for materials simulations, *Reviews of Modern Physics* **96**, 045008 (2024).
- [71] H. Mori, T. Nomoto, R. Arita, and E. R. Margine, Efficient anisotropic Migdal-Eliashberg calculations with an intermediate representation basis and Wannier interpolation, *Physical Review B* **110**, 064505 (2024).
- [72] K. Momma and F. Izumi, *VESTA3* for three-dimensional visualization of crystal, volumetric and morphology data, *Journal of Applied Crystallography* **44**, 1272 (2011).
- [73] M. Kawamura, FermiSurfer: Fermi-surface viewer providing multiple representation schemes, *Computer Physics Communications* **239**, 197 (2019).
- [74] A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder, and K. A. Persson, Commentary: The materials project: A materials genome approach to accelerating materials innovation, *APL Materials* **1**, 011002 (2013).
- [75] B. Albert and K. Schmitt, CaB_2C_2 : Reinvestigation of a semiconducting boride carbide with a layered structure and an interesting boron/carbon ordering scheme, *Inorganic Chemistry* **38**, 6159 (1999).
- [76] Z. Yuan, W. Hu, and D. Yu, High-pressure synthesis, crystal structure, and physical properties of $\text{NaB}_{13}\text{C}_2$ single crystals, *Journal of Alloys and Compounds* **893**, 162320 (2022).
- [77] Y. Zhang, Advancing DFT predictions in Cu-chalcogenides with full-yet-shallow 3d-orbitals: Meta-GGA plus Hubbard-like U correction, *The Journal of Chemical Physics* **161**, 174109 (2024).
- [78] G. Fabbris, A. Thorn, W. Bi, M. Abramchuk, F. Bahrami, J. H. Kim, T. Shinmei, T. Irifune, F. Tafti, A. N. Kolmogorov, and D. Haskel, Complex pressure-temperature structural phase diagram of the honeycomb iridate Cu_2IrO_3 , *Physical Review B* **104**, 014102 (2021).
- [79] T. C. King, P. D. Matthews, H. Glass, J. A. Cormack, J. P. Holgado, M. Leskes, J. M. Griffin, O. A. Scherman, P. D. Barker, C. P. Grey, S. E. Dutton, R. M. Lambert, G. Tustin, A. Alavi, and D. S. Wright, Theory and practice: Bulk synthesis of c3b and its h2- and li-storage capacity, *Angewandte Chemie International Edition* **54**, 5919 (2015).
- [80] T. Björkman, A. Gulans, A. V. Krashenninnikov, and R. M. Nieminen, van der Waals bonding in layered compounds from advanced density-functional first-principles calculations, *Physical Review Letters* **108**, 235502 (2012).
- [81] D. Campi, N. Mounet, M. Gibertini, G. Pizzi, and N. Marzari, Expansion of the Materials Cloud 2D Database, *ACS Nano* **17**, 11268 (2023).
- [82] A. K. Verma, P. Modak, D. M. Gaitonde, R. S. Rao, B. K. Godwal, and L. C. Gupta, Possible High-Temperature Superconductivity in Hole-Doped MgB_2C_2 , *Europhysics Letters* **63**, 743 (2003).
- [83] P. B. Allen and R. C. Dynes, Transition temperature of strong-coupled superconductors reanalyzed, *Phys. Rev. B* **12**, 905 (1975).
- [84] S. B. Mishra, E. T. Marcial, S. Debata, A. N. Kolmogorov, and E. R. Margine, Stability-superconductivity map for compressed Na-intercalated graphite, *Physical Review B* **110**, 174508 (2024).
- [85] N. K. Nepal and L.-L. Wang, Machine-learning Guided Search for Phonon-mediated Superconductivity in Boron and Carbon Compounds, *arXiv preprint arXiv:2409.18441* (2024).
- [86] J. M. An and W. E. Pickett, Superconductivity of MgB_2 : Covalent Bonds Driven Metallic, *Physical Review Letters* **86**, 4366 (2001).
- [87] J. Kortus, I. I. Mazin, K. D. Belashchenko, V. P. Antropov, and L. L. Boyer, Superconductivity of Metallic Boron in MgB_2 , *Physical Review Letters* **86**, 4656 (2001).
- [88] M. Calandra and F. Mauri, Theoretical Explanation of Superconductivity in C_6Ca , *Physical Review Letters* **95**, 237002 (2005).
- [89] I. I. Mazin, Intercalant-Driven Superconductivity in YbC_6 and CaC_6 , *Physical Review Letters* **95**, 227001 (2005).
- [90] A. Y. Liu and I. I. Mazin, Combining the advantages of superconducting MgB_2 and CaC_6 in one material: Suggestions from first-principles calculations, *Physical Review B* **75**, 064510 (2007).
- [91] M. Calandra, A. N. Kolmogorov, and S. Curtarolo, Search for high T_c in layered structures: The case of LiB , *Physical Review B* **75**, 144506 (2007).
- [92] A. Sanna, G. Profeta, A. Floris, A. Marini, E. K. U. Gross, and S. Massidda, Anisotropic gap of superconducting CaC_6 : A first-principles density functional calculation, *Physical Review B* **75**, 020511 (2007).
- [93] P. C. Canfield and G. W. Crabtree, Magnesium Diboride: Better Late than Never, *Physics Today* **56**, 34 (2003).
- [94] S.-L. Yang, J. A. Sobota, C. A. Howard, C. J. Pickard, M. Hashimoto, D. H. Lu, S.-K. Mo, P. S. Kirchmann, and Z.-X. Shen, Superconducting graphene sheets in CaC_6 enabled by phonon-mediated interband interactions, *Nature Communications* **5**, 3493 (2014).
- [95] C.-M. Hao, X. Li, A. R. Oganov, J. Hou, S. Ding, Y. Ge, L. Wang, X. Dong, H.-T. Wang, G. Yang, X.-F. Zhou, and Y. Tian, Superconductivity in compounds of sodium-intercalated graphite, *Physical Review B* **108**, 214507 (2023).
- [96] N. Geng, K. P. Hilleke, L. Zhu, X. Wang, T. A. Strobel, and E. Zurek, Conventional High-Temperature Superconductivity in Metallic, Covalently Bonded, Binary-Guest C-B Clathrates, *Journal of the American Chemical Society* **145**, 1696 (2023).
- [97] W. Hayami, X. Rocquefelte, and J. F. Halet, Possible Su-



perconductivity for Layered Metal Boride Carbide Compounds MB_2C_2 (M = Alkali, Alkaline-Earth, or Rare-Earth Metals), *Inorganic Chemistry* **63**, 20975–20983 (2024).

- [98] J. Towns, T. Cockerill, M. Dahan, I. Foster, K. Gaither, A. Grimshaw, V. Hazlewood, S. Lathrop, D. Lifka, G. D. Peterson, R. Roskies, J. R. Scott, and N. Wilkins-Diehr,

XSEDE: Accelerating scientific discovery, *Computing in Science & Engineering* **16**, 62 (2014).

- [99] D. Stanzione, J. West, R. T. Evans, T. Minyard, O. Ghattas, and D. K. Panda, Frontera: The evolution of leadership computing at the national science foundation, in *Practice and Experience in Advanced Research Computing* (ACM, New York, NY, USA, 2020) pp. 106–111.



The data supporting the findings of this study are available within the article and its Supplementary Information file. Additional data are available from the corresponding author upon request.

