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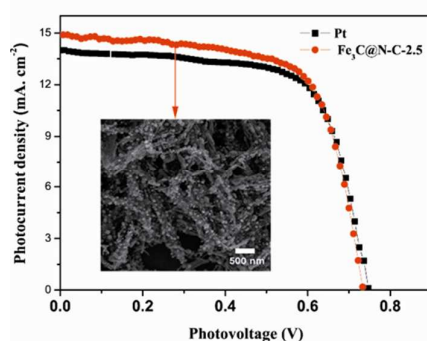
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Graphical Abstract

The hierarchical nanocomposites ($\text{Fe}_3\text{C@N-C}$) of iron carbide encaged in nitrogen-doped carbon counter electrode was fabricated for low-cost, highly efficient dye-sensitized solar cells.



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ARTICLE TYPE

Nitrogen-doped carbon and iron carbide nanocomposites as cost-effective counter electrodes of dye-sensitized solar cells

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The hierarchical nanocomposites of iron carbide (Fe_3C) encaged in nitrogen-doped carbon (N-C) were prepared by using a simple carbothermal reduction of iron(II) oxalate (FeC_2O_4) nanowires in the presence of cyanamide (NH_2CN) at 600 °C. Such $\text{Fe}_3\text{C}@$ N-C nanocomposites delivered fair electrocatalytic activity for I_3^-/I^- redox reaction. As a result, to be explored as cost-effective counter electrodes of dye-sensitized solar cells, an efficiency of 7.36% was achieved, which was comparable to that of the cell with Pt-FTO counter electrode (7.15%) under the same experimental condition. The good electrochemical performance is attributed to synergistic effect of the combination of N-C and Fe_3C and one dimensional configuration, which endow the nanocomposites more interfacial active sites and improved electron transfer efficiency for the reduction of I_3^-/I^- .

Introduction

Nanocrystalline dye-sensitized solar cells (DSSCs), introduced by Grätzel and O'Regan in 1991, have attracted considerable attention from academic and industrial field because of cost effectiveness and comparable high light-to-electricity conversion efficiency (η) and ease of manufacturing.¹⁻⁶ The counter electrodes (CE), as one of essential components of the DSSCs, are usually composed of the noble metal platinum (Pt) coated on the transparent conductive oxides (such as indium-doped tin oxide). However, the limited supply and the high cost of Pt hinder their large-scale commercial applications in DSSCs counter electrode.^{7,8} Therefore, it is imperative to develop low-cost, abundant and highly-efficient substitutes for the conventional Pt counter electrode in the DSSCs system. Many materials such as carbon materials,^{5,9-12} conductive polymers,^{13,14} inorganic materials including metal nitrides,¹⁵⁻¹⁷ carbides,^{18,19} sulfides²⁰⁻²³ and selenide^{24,25} and composites have been employed to replace Pt as CE in DSSCs.

Among these counter electrode materials, transition metal carbides, such as Molybdenum, Tungsten, Niobium and Vanadium,^{18,19} have been shown to have good catalytic activity for the reduction of triiodide to iodide in DSSCs. However, these metal carbides contain rare earth elements. Iron carbide (Fe_3C) seems more attractive for its abundance in the earth's crust and its high electronic conduction, higher resistance against oxidation and good catalytic activity²⁶ as well. However, iron carbide (Fe_3C) has been less investigated in DSSCs mainly because of the preconception that iron carbide is metastable compound, which readily decomposes into α -Fe and carbon.²⁷ Moreover, α -Fe is susceptible to corrosion by I^-/I_3^- redox species.²⁸ Very recently, Fu *et al.* found that the Fe_3C could survive very well in I^-/I_3^-

electrolyte and fair efficiency of 6.04% was obtained. However, as mentioned in this work, α -Fe is coexisted in the compound and an additional etching process is required to remove the unstable α -Fe.²⁹

As the counter electrode of DSSCs, the electronic conductivity is as equally important as the catalytic activity to decrease the overvoltage to minimize the energy losses.³⁰ It is known that catalytic and electronic properties of transition metal compounds are governed by their intrinsic materials. However, Bulk materials usually exhibit limited catalytic activity probably because of their large particle size and less specific surface area.³¹ Nanostructure materials have been known to show better properties compared to their corresponding bulk materials. Therefore, it is great significance to further explore high efficient counter electrode of transition metal compounds by rational design of nanostructured catalytic materials. Among different kinds of nanoscale morphologies, one-dimensional nanostructure is known to be beneficial for electronic conduction along the axial direction.²² Herein, iron carbide encaged in nitrogen-doped carbon ($\text{Fe}_3\text{C}@$ N-C) nanowires were synthesized by directly annealing of iron(II) oxalate (FeC_2O_4) nanowires in H_2 atmosphere with cyanamide (NH_2CN) as the structure confinement agent. At optimized annealing temperature, the composites with varied Fe_3C concentration were explored as the counter electrode in DSSCs. Because of synergetic effect of N-C and Fe_3C , and the presence of one dimensional morphology, $\text{Fe}_3\text{C}@$ N-C-2.5 (ratio of $\text{FeC}_2\text{O}_4/\text{CH}_2\text{N}_2$ is 2.5) nanostructures as counter electrodes exhibited comparable photovoltaic performance compared with those of Pt-FTO.

Experimental

Materials

Cetyltrimethylammonium bromide (CTAB), Iron (II) chloride tetrahydrate ($\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$), and oxalic acid ($\text{H}_2\text{C}_2\text{O}_4$) were purchased from Sinopharm Chemical Reagent Co., Ltd., China. Iodine and 4-*tert*-butylpyridine were purchased from TCI. N719 dye ($(\text{Ru}(\text{dcbpy})_2(\text{NCS})_2$, (dcbpy = 2,2-bipyridyl-4,4-dicarboxylate)) was purchased from Solaronix SA. TiO_2 paste and 1,2-dimethyl-3-propylimidazolium iodide were purchased from Wuhan GeAo Tech Co., Ltd., China. Guanidinium thiocyanate was purchased from Dalian HeptaChroma Solar Tech Co., Ltd., China. All the chemicals are analytical grade without further purification.

Synthesis of $\text{Fe}_3\text{C}@\text{N-C}$

FeC_2O_4 nanowires were firstly synthesized as precursor of Fe_3C product via a microemulsion route. 10 g CTAB which served as soft template was added to a mixture of cyclohexane (300 mL) and *n*-pentanol (10 mL). After stirring for 30 min, 15 mL of 1 M $\text{H}_2\text{C}_2\text{O}_4$ and 5 mL of 1 M $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$ aqueous solution was added to the above microemulsion and stirred for another 24 hrs at ambient temperature. The yellow precipitate was filtered and washed with ethanol to remove the impurity. After dried overnight in a vacuum oven at 50°C , the as-synthesized FeC_2O_4 nanowires was dispersed in ethanol which contained a certain amount of NH_2CN for homogenous precursor mixing before heat treatment. After dried again, the mixed raw powders were annealed at various temperatures for 2 hrs under hydrogen atmosphere and the black $\text{Fe}_3\text{C}@\text{N-C}$ nanocomposites were finally obtained. The obtained $\text{Fe}_3\text{C}@\text{N-C}$ nanocomposites with the weight ratio of $\text{FeC}_2\text{O}_4/\text{NH}_2\text{CN}$ of 1, 2.5, 4 were labeled as $\text{Fe}_3\text{C}@\text{N-C-1}$, $\text{Fe}_3\text{C}@\text{N-C-2.5}$, $\text{Fe}_3\text{C}@\text{N-C-4}$.

Preparation of counter electrodes

The mirror-like Pt/FTO electrode was obtained by electrodeposition a platinum layer on the surface of fluorine-doped tin oxide substrate. The thickness of Pt films is about 75 nm. The $\text{Fe}_3\text{C}@\text{N-C}$ (30 mg) and polyvinylidene fluoride dissolved in *N*-methyl-2-pyrrolidinone (10%, wt.%) were grinded together to generate homogenous paste. Subsequently, the counter electrode films were prepared on precleaned fluorine-doped tin oxide (FTO) substrate by doctor blade technique followed by heat drying at 60°C for 24 hrs.

Fabrication of DSSCs

TiO_2 working photoanodes were prepared on FTO substrate using TiO_2 pastes by doctor blade technique and subsequently sintered at 500°C for 30 min in air. The resultant TiO_2 photoanodes were soaked in an ethanol solution of N719 dye (3×10^{-4} M) for 24 hrs to obtain dye-sensitized TiO_2 electrode. The dye-adsorbed TiO_2 photoanodes with an active area of 0.16 cm^2 were assembled with $\text{Fe}_3\text{C}@\text{N-C}$ and platinum counter electrodes using laboratory tape as a spacer to fabricate corresponding sandwich-type cells, respectively. The liquid electrolyte is composed of 0.6 M 1,2-dimethyl-3-propylimidazolium iodide (DMPII), 0.03 M iodine (I_2), 0.06 M lithium iodide (LiI), 0.5 M 4-*tert*-butylpyridine (TBP), and 0.1 M guanidinium thiocyanate with acetonitrile (ACN) as the solvent.

Characterization

The morphologies of $\text{Fe}_3\text{C}@\text{N-C}$ were investigated using field

emission scanning electron microscopy (FESEM, HITACHI S-4800), and high-resolution transmission electron microscopy (HRTEM, JEOL 2010F). X-ray diffraction (XRD) patterns were recorded with a Bruker-AXS Micro-diffractometer (D8 ADVANCE) using $\text{Cu K}\alpha$ radiation ($\lambda = 1.5406 \text{ \AA}$) from 20°C to 70°C . Cyclic voltammetry (CV) was carried out in a three-electrode system in an acetonitrile solution of 0.1 M LiClO_4 , 10 mM LiI, and 1 mM I_2 . Platinum served as a counter electrode and the non-aqueous Ag/Ag^+ couple was applied as a reference electrode. The photocurrent-voltage characteristics of the DSSCs were measured with a Newport (USA) solar simulator (300 W Xe source) and a Keithley 2440 source meter. Electrochemical impedance spectroscopy (EIS) measurements were performed using a Zahner Zennium electrochemical workstation by applying an AC voltage of 10 mV amplitude in the frequency range between 100 kHz and 100 mHz at room temperature. Fitting of impedance spectra to the proposed equivalent circuit was performed by using the Zview software.

Result and discussion

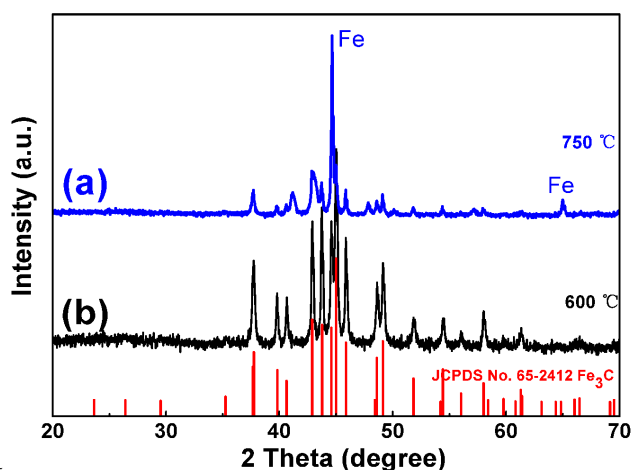


Fig. 1 Typical XRD patterns of Fe_3C samples synthesized at 750°C and 600°C .

Fig. 1 depicts the XRD patterns of Fe_3C samples ($\text{Fe}_3\text{C}@\text{N-C-2.5}$) synthesized at 600°C and 750°C when the weight ratio of $\text{FeC}_2\text{O}_4/\text{NH}_2\text{CN}$ is 2.5 ($R = 2.5$). After annealed at 600°C , all the peaks of the sample could be indexed to orthorhombic Fe_3C (JCPDS No. 65-2412) without any impurity phase which has been shown in Fig. 1(b). However, when the reaction temperature was raised to 750°C , two peaks corresponding to $\alpha\text{-Fe}$ were detected and shown in Fig. 1(a). This is consistent with the previously reported results that the decomposition of Fe_3C into $\alpha\text{-Fe}$ and carbon occurred at $T > 600^\circ\text{C}$.²⁷ No diffraction peaks of graphite were observed which implied that the carbon component in the composites was amorphous due to the low annealed temperature. In order to avoid the generation of $\alpha\text{-Fe}$ which suffers from chemical instability in I/I_3^- electrolyte, the optimized reaction temperature was 600°C . The influence of $\text{FeC}_2\text{O}_4/\text{NH}_2\text{CN}$ ratio on product composition was also evaluated at the optimized temperature. As shown in Fig. S1, both the patterns exhibited Fe_3C phase which coincide with the sample obtained at $R = 2.5$ (Fig. 1(b)). This implied that the $\text{FeC}_2\text{O}_4/\text{NH}_2\text{CN}$ ratio may only change the relative amount of

Fe₃C and carbon but does not affect the crystal structure of Fe₃C in final product. The element analysis results shown in Table S1 indicated as-synthesized Fe₃C sample comprised of C, N, Fe. It is reported that NH₂CN was condensed to C₃N₄ and decomposed into N-doped carbon(C-N) at high temperature.³³ Therefore, Fe₃C sample was the composite of Fe₃C and minor amount of N-doped carbon. Meanwhile, the element analysis results indicated that with an increasing FeC₂O₄/NH₂CN ratio, the content of Fe₃C increased correspondingly in the as-prepared product.

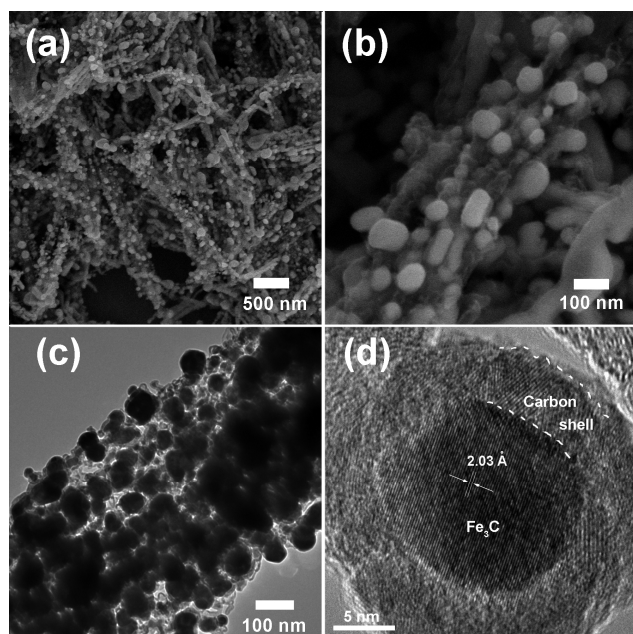


Fig. 2 Typical SEM and its enlarged images (a, b), TEM (c) and HRTEM (d) images of Fe₃C@N-C-2.5.

Typical SEM and TEM images of as-synthesized FeC₂O₄ are shown in Fig. S2. It can be clearly seen that the obtained FeC₂O₄ exhibits one dimensional nanowire morphology and relatively smooth surface. After reacted with NH₂CN (R=2.5), chain-like Fe₃C@N-C nanomaterials were observed in Fig. 2(a) and most of the composites maintained the one dimensional morphology. This may be due to the appropriate amount of NH₂CN (R = 2.5) addition which could prevent nanowires from aggregation. When the FeC₂O₄/NH₂CN ratio was varied to 1 as shown in Fig. S3 (a) and (b), most of the Fe₃C nanowires displayed a rod-like aggregation structure which may be unfavorable to electronic transportation. When the FeC₂O₄/NH₂CN ratio was raised to 4, Fe₃C with minor carbon coating were observed which are shown in Fig. S3 (c) and (d), where well electronic-connection is difficult to be achieved between isolated Fe₃C nanoparticles. However, from the enlarged micrographs shown in Fig. 2(b) and 2(c), Fe₃C nanoparticles with size below 100 nm were well connected together by N-doped carbon at R=2.5. This unique structure is expected to be beneficial to the electron transfer during catalytic reaction. HRTEM given in Fig. 2 (d) confirmed that Fe₃C nanoparticle was well encapsulated by a N-doped carbon layer with the fringe spacing of 2.03 Å corresponding to the (112) planes of orthorhombic Fe₃C.³⁴ In addition, elemental mapping in Fig. S4 elucidated that Fe, C and N were homogeneously distributed in composites. The much narrower D and G peak width in Fig. S5 implied that the carbon of Fe₃C@N-C-2.5 was

more crystalline than that of Fe₃C@N-C-1. From the above results, well-wired Fe₃C nanoparticles, as well as the carbon sheath in Fe₃C@N-Cnanocomposite may expect to present wonderful catalytic activity when used as CE in DSSCs.

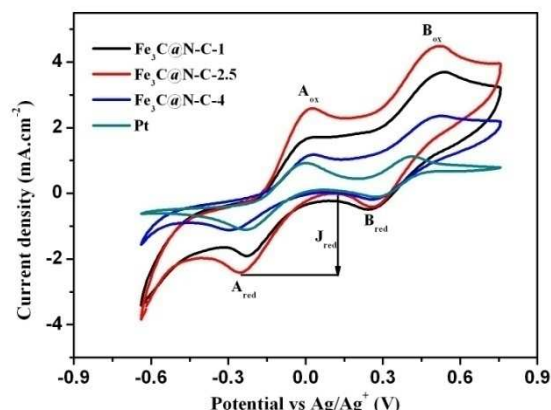


Fig. 3 Cyclic voltammograms of Pt, Fe₃C@N-C counter electrodes with different raw materials ratio in 10 mM LiI, 1 mM I₂ and 0.1 M LiClO₄ acetonitrile solution at a scan rate of 20 mV s⁻¹.

Cyclic voltammetry (CV) was performed to evaluate the electrocatalytic activity of counter electrodes to reduce triiodide using a three-electrode system. CV curves for I₃⁻/I⁻ redox reaction obtained on Fe₃C@N-C and the reference Pt counter electrodes at a scan rate of 20 mV s⁻¹ are shown in Fig. 3. Counter electrodes with good electrocatalytic activity for I₃⁻/I⁻ electrolyte display two typical pairs of redox peaks, A_{ox}/A_{red} and B_{ox}/B_{red}. The A_{ox}/A_{red} is assigned to the redox reaction shown in Eq. (1) and B_{ox}/B_{red} is ascribed to redox reaction shown in Eq. (2).³⁵ From Fig. 3, it can be seen the CV curves of the CEs with Fe₃C@N-C displayed two pairs of redox peaks, indicating that they all possess electrocatalytic ability for the reduction of triiodide ions. A higher reduction peak current density (J_{red}) and a lower peak-to-peak voltage separation (V_{pp}) indicate a better catalytic activity. The redox peaks (A_{ox} and A_{red}) directly affects the DSSC performance. Therefore, we focused on investigation about peak current density (J_{red}) and peak-to-peak voltage separation (V_{pp}) of A_{ox} and A_{red}. The profile and location of the A_{ox} and A_{red} redox peaks of Fe₃C@N-C CEs were close to those of Pt CE. This indicated Fe₃C@N-C CEs and Pt CE possessed similar E_{pp} value. However, the cathodic current density increased in the order of Fe₃C@N-C-4 (1.15 mA cm⁻²) < Pt(1.21 mA cm⁻²) < Fe₃C@N-C-1 (1.68 mA cm⁻²) < Fe₃C@N-C-2.5 (2.41 mA cm⁻²). By comprehensive consideration of the E_{pp} and peak current density, the Fe₃C@N-C-2.5 presented slightly closer E_{pp} but much larger cathodic current density than that of the Pt CE, demonstrating a relatively better electrocatalytic activity than that of the Pt CE. The enhanced electrocatalytic activity can be attributed to synergistic effect of the combination of high catalytic activity and good electrical conductivity of nitrogen-doped carbon into Fe₃C and the morphology of nanowires which is beneficial to electron transfer.³⁶ Fig. S6 shows a 100-cycles CVs for Fe₃C@N-C-2.5 electrode at a scan rate of 20 mV s⁻¹. They present almost unchanged curves shape and constant redox peak current densities, indicating an comparable electrochemical stability of the Fe₃C@N-C-2.5 CE.

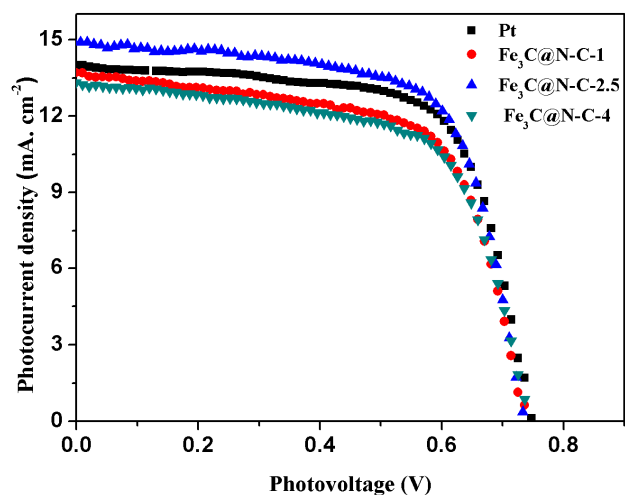
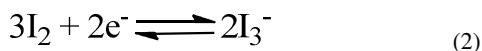
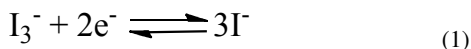


Fig. 4 Characteristic photocurrent density-voltage (J - V) curves of DSSCs with different electrodes, measured under simulated sunlight 100 mW cm^{-2} (AM 1.5). The liquid electrolyte is composed of 0.6 M 1,2-dimethyl-3-propylimidazolium iodide (DMPII), 0.03 M iodine (I_2), 0.06 M lithium iodide (LiI), 0.5 M 4-*tert*-butylpyridine (TBP), and 0.1 M guanidinium thiocyanate in acetonitrile solution.

The photocurrent density-voltage (J - V) characteristic curves of the DSSCs fabricated with different counter electrodes measured under the illumination of 1 sun (100 mW cm^{-2}) are shown in Fig. 4. The photovoltaic parameters of these devices, including the short-circuit current (J_{sc}), the open-circuit voltage (V_{oc}), the fill factor (FF), and the energy conversion efficiency (η), are summarized in Table 1. All devices, with the CEs containing Fe_3C , had similar V_{oc} . However, J_{sc} (FF) complied with the order $\text{Fe}_3\text{C@N-C-2.5} > \text{Pt} > \text{Fe}_3\text{C@N-C-1} > \text{Fe}_3\text{C@N-C-4}$. This is in accordance with the results of CVs measurements. The DSSCs with $\text{Fe}_3\text{C@N-C}$ counter electrodes exhibited a relatively higher η . The $\text{Fe}_3\text{C@N-C-1}$, $\text{Fe}_3\text{C@N-C-2.5}$, and $\text{Fe}_3\text{C@N-C-4}$ give η of 6.52%, 7.36%, and 6.36%, respectively. The DSSCs with $\text{Fe}_3\text{C@N-C-2.5}$ CE possessed the best energy conversion efficiencies of 7.36% which was comparable with that of the DSSCs with Pt CE (7.15%).

Table 1 Characteristics of the J - V curves of the DSSCs fabricated using different counter electrodes

Counter Electrode	J_{sc} (mA/cm ²)	V_{oc} (mV)	FF (%)	η (%)
$\text{Fe}_3\text{C@N-C-1}$	13.74	738	64.31	6.52 ± 0.04
$\text{Fe}_3\text{C@N-C-2.5}$	14.97	741	66.35	7.36 ± 0.03
$\text{Fe}_3\text{C@N-C-4}$	13.39	741	64.08	6.36 ± 0.04
Pt	14.13	747	67.71	7.15 ± 0.02

^a V_{oc} : open circuit voltage, J_{sc} : short circuit current density, FF: fill factor, η : energy conversion efficiency.

In order to further evaluate the electrochemical activity of the composite materials as counter electrodes in DSSCs, the electrochemical impedance spectra (EIS) were measured in a symmetric sandwich cell configuration consisting of two identical counter electrodes. Their Nyquist plots are illustrated in Fig. 5. For comparison, the impedance spectrum of the cell consisting of the conventional platinized electrodes was also presented here.

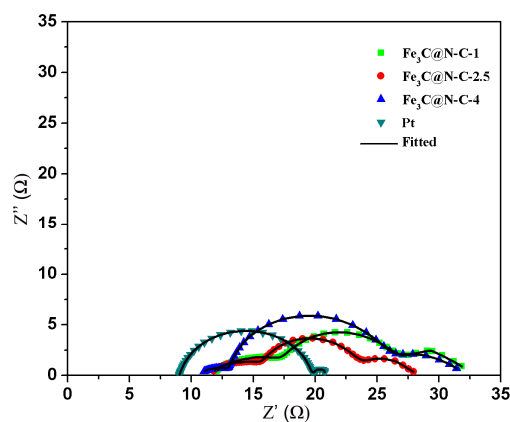


Fig. 5 Nyquist plots for the symmetric cells fabricated with two identical counter electrodes of $\text{Fe}_3\text{C@N-C-1}$ (■), $\text{Fe}_3\text{C@N-C-2.5}$ (●), $\text{Fe}_3\text{C@N-C-4}$ (▲), Pt (▼). The lines express fit results for corresponding EIS data. The cells were measured with the frequency range between 100 kHz and 100 mHz.

Table 2 EIS parameters of the symmetric cells based on different counter electrodes

Counter Electrode	R_s (Ω)	R_{ct} (Ω)	$R_{ct}(S)$ (Ω)
$\text{Fe}_3\text{C@N-C-1}$	10.9	7.77	7.10
$\text{Fe}_3\text{C@N-C-2.5}$	10.75	6.78	5.58
$\text{Fe}_3\text{C@N-C-4}$	10.38	12.3	2.83
Pt	9.04	10.65	~

^a R_s : series resistance, R_{ct} : charge-transfer resistance, $R_{ct}(S)$: impedance arising from the $\text{Fe}_3\text{C@N-C-FTO}$ interface

For a conventional symmetric cell consisting of platinized electrodes, the electric circuit elements should have a series resistance (R_s), a constant phase element (CPE), the charge transfer resistance (R_{ct}) and Nernst diffusion impedance of the I_3^-/I^- redox species within a thin layer in the electrolyte.³⁷ However, for a symmetric cell based on $\text{Fe}_3\text{C@N-C}$ electrodes, three semicircles were visible for $\text{Fe}_3\text{C@N-C}$ compared with two semicircle of the conventional Pt. The semicircle in the high frequency region was speculated to correspond to impedance arising from the $\text{Fe}_3\text{C@N-C-FTO}$ interface ($R_{ct}(S)$).³⁸ The one in the middle frequency region is associated with the charge-transfer resistance of counter electrode/redox (I/I_3^-) interface and the capacitance of the counter electrode/electrolyte interface. The low-frequency semicircle is attributed to Nernst diffusion impedance of the I_3^-/I^- redox species within a thin layer in the electrolyte. The intercept of the real axis at high frequency represents the ohmic series resistance including the sheet resistance of two identical CEs and the electrolytic resistance. The equivalent circuit is given in the Fig. S7 and the simulated data from the EIS spectra for $\text{Fe}_3\text{C@N-C}$ and Pt are summarized in Table 2. The $\text{Fe}_3\text{C@N-C}$ CEs had slightly larger R_s than Pt CE (Table 2), indicating that $\text{Fe}_3\text{C@N-C}$ had relatively lower conductivity compared with Pt. The impedance arising from the $\text{Fe}_3\text{C@N-C-FTO}$ interface of $\text{Fe}_3\text{C@N-C-1}$, $\text{Fe}_3\text{C@N-C-2.5}$ and $\text{Fe}_3\text{C@N-C-4}$ are 7.10 Ω, 5.58 Ω, and 2.83 Ω. The results demonstrated that bonding strength between $\text{Fe}_3\text{C@N-C}$ and FTO became stronger with the increase of content of Fe_3C . The simulated charge-transfer resistances of $\text{Fe}_3\text{C@N-C-1}$, $\text{Fe}_3\text{C@N-C-2.5}$, and $\text{Fe}_3\text{C@N-C-4}$ counter electrode are 7.77 Ω, 6.78 Ω, and 12.3 Ω respectively. The charge-transfer resistances

Fe₃C@N-C-1 and Fe₃C@N-C-2.5 are much lower than that of Pt electrode (10.65 Ω), suggesting the higher electrocatalytic activity of Fe₃C@N-C electrodes over Pt-FTO electrode for the reduction of triiodide ions. The charge-transfer resistance of Fe₃C@N-C-2.5 is much lower than that of other Fe₃C@N-C electrode which may be attributed to efficient electrocatalytic activity from the optimal interaction between the Fe₃C and carbon and enhanced electron transport capability contributed by the nanowire morphology.³⁶ The enhanced charge-transfer resistance of Fe₃C@N-C-4 may be due to decreased content of carbon which decreased electrocatalytic activity of Fe₃C@N-C composites. The EIS results agree with the CV data. The lower resistance would endow a greater FF and higher η in solar cell which is corrugated by the corresponding performance measurement.

Conclusions

In summary, N-C and Fe₃C nanocomposite (Fe₃C@N-C) was fabricated by carbothermal reduction using cyanamide as nitrogen and carbon source. Appropriate amount of NH₂CN prevented FeC₂O₄ nanowires aggregation and generated a favorable carbon-coating at 600 °C. Dye-sensitized solar cells with Fe₃C@N-C nanocomposite films as the counter electrode were explored. Among these DSSCs fabricated from Fe₃C@N-C, Fe₃C@N-C-2.5 yielded the highest photoelectrical conversion efficiency of 7.36%. That is because that the synergetic combination of the N-doped carbon and Fe₃C, generating better catalytic performance and lowest charge-transfer resistance. Moreover, Fe₃C@N-C-2.5 nanowire configuration is favorable for fast electron transfer. The abundance of Fe element and facile synthesis method endowed Fe₃C-based nanocomposites promising candidate of large-scale, highly efficient and low-cost counter electrode for DSSCs.

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Notes and references

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[†] Electronic Supplementary Information (ESI) available: XRD patterns of Fe₃C@N-C nanocomposites with different FeC₂O₄/NH₂CN ratio, Element analysis results of Fe₃C@N-C with different R value, SEM and TEM images of FeC₂O₄, TEM images of Fe₃C@N-C-1 and Fe₃C@N-C-4, Elemental mapping of Fe₃C@N-C nanocomposites, Raman spectra of Fe₃C@N-C-1, Fe₃C@N-C-2.5, Consecutive 100 cyclic voltammograms for the Fe₃C@N-C-2.5 CE, equivalent circuits for the symmetric cells consisted of platinum electrodes (a) Fe₃C@N-C (b), SEM images of

- nearly pure N-C and non-1D configuration Fe₃C@N-C-2.5 and Characteristics of the J-V curves of the DSSCs fabricated using nearly pure N-C and non-1D configuration Fe₃C@N-C-2.5. See DOI: 10.1039/b000000x/
[‡] These authors contributed equally to this work.
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